

0.1 X-ray Diffraction

0.1.1 Scattering at Lattices

To elucidate the working principles behind X-ray diffraction (XRD) as a measurement method, a brief description of reciprocal space and constructive interference will be provided. Those derivations are based on Ashcroft and Mermin (1976) [1].

A periodic point-like structure with translational symmetry (“BRAVAIS lattice”) can be described by three vectors \mathbf{a}_i that span a so-called “unit cell”. Every lattice point \mathbf{R} is a linear combination of those unit cell vectors. For such a lattice, there exists a so-called “reciprocal lattice”, which consists of all vectors \mathbf{K} satisfying the condition¹:

$$e^{i\langle \mathbf{K}, \mathbf{R} \rangle} = 1. \quad (1)$$

This is again a BRAVAIS lattice with unit cell vectors \mathbf{a}_j^* :

$$\mathbf{K}_{hkl} = h\mathbf{a}_1^* + k\mathbf{a}_2^* + l\mathbf{a}_3^*. \quad (2)$$

It follows that for any i, j :

$$\langle \mathbf{a}_i^*, \mathbf{a}_j \rangle = 2\pi\delta_{ij}, \quad (3)$$

with the KRONECKER delta δ_{ij} . A major application of reciprocal space vectors is their ability to describe lattice planes. Any lattice plane can be described by the shortest possible reciprocal space vector \mathbf{K}_{hkl} perpendicular to it. Consequently, the lattice plane is denoted by (hkl) . The distance between equivalent lattice planes can be calculated via $d_{hkl} = |\mathbf{K}_{hkl}|^{-1}$.

With those preliminaries, the conditions for constructive interference during diffraction of radiation at BRAVAIS lattices can be derived. Consider two scattering centers separated by \mathbf{d} . Now consider incoming radiation with wave vector \mathbf{k} :

$$\mathbf{k} = \frac{2\pi}{\lambda} \hat{\mathbf{n}}, \quad (4)$$

with wavelength λ and direction $\hat{\mathbf{n}}$. For the case of elastic scattering, the outgoing wave vector \mathbf{k}' has the same wavelength λ but different direction $\hat{\mathbf{n}}'$. The phase difference of two photons scattered at the 1st and 2nd scattering center, respectively, can be calculated from their path difference, which reads

$$\langle \mathbf{d}, \hat{\mathbf{n}} \rangle + \langle -\mathbf{d}, \hat{\mathbf{n}}' \rangle. \quad (5)$$

Constructive interference occurs, if the phase difference is an integral multiple of the wavelength, so it must follow that

$$\langle \mathbf{d}, (\hat{\mathbf{n}} - \hat{\mathbf{n}}') \rangle = m\lambda \quad (6)$$

$$\Leftrightarrow \langle \mathbf{d}, (\hat{\mathbf{k}} - \hat{\mathbf{k}}') \rangle = 2\pi m, \quad (7)$$

with $m \in \mathbb{N}$. Comparing with (1) reveals that $\hat{\mathbf{k}} - \hat{\mathbf{k}}'$ is a reciprocal space vector, because the separation \mathbf{d} of the two scattering centers is a lattice vector. So constructive interference (observing a reflex) occurs if and only if the scattering geometry (determined by

¹The definition of \mathbf{K} by (1) is a consequence of demanding that the plane wave described by $f_{\mathbf{K}}(\mathbf{r}) = \exp(i\langle \mathbf{K}, \mathbf{r} \rangle)$ has the same symmetry as the BRAVAIS lattice [1].

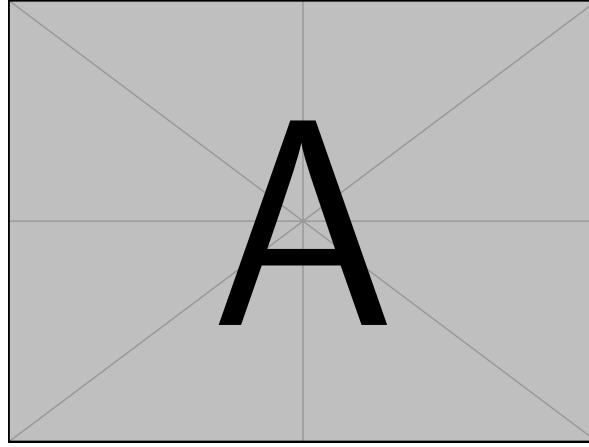


Figure 1: Here comes an ashcroft-like image for the scattering geometry to derive Equ. (5)

angle of incidence and refraction, as well as wavelength) matches the lattice symmetry in the sense that there is a corresponding lattice translation vector \mathbf{d} fulfilling Equ. (7). So from the “position” of reflexes, one can deduce the lattice symmetry.

Note that this description of X-ray scattering is equivalent to the BRAGG condition:

$$m\lambda = 2d_{hkl} \sin(\theta), \quad (8)$$

where the angle of incidence θ and λ are contained in $\hat{\mathbf{k}} - \hat{\mathbf{k}}'$. Furthermore, when a lattice point is not equivalent to a single atom, but represents several scattering centers, an additional geometrical structure factor has to be taken into account to determine whether a certain geometry allows reflexes. This is important, e.g., for structures with trigonal symmetry. They are described with a conventional hexagonal unit cell, although not every plane (hkl) exhibits constructive interference.

0.1.2 X-rays

Atomic distances in solids are of the order of several Å, so the radiation for probing those structures must have a similar wavelength, which turns out to be X-rays [2]. The following description of X-rays is based on Spieß (2009) [3].

The basis of any X-ray tube are high-energy electrons which are produced by thermionic emission in a cathode, which is usually made out of tungsten². An electric field of several kV accelerates the electrons to the anode, where they are stopped such that around 99 % of their kinetic energy dissipates. The momentum change of electrons, which are charged particles, leads to emission of *bremsstrahlung*. Furthermore, the electrons ionize atoms of the anode material which leads to unoccupied electron states. If those states are filled by electrons with higher quantum number n , the difference in energy of those levels is emitted as radiation with a discrete spectrum, called characteristic X-ray. Important for this work is a part of the characteristic spectrum, called K-radiation, which originates in occupation of empty 1s-orbitals. The occupying electron must come from an orbital with angular momentum quantum number $l = 1$, i.e.

²Tungsten is the element with the second highest melting point of around 3400 °C. This ensures a low contamination of the anode with cathode material.

a p -orbital, because Δl cannot be zero. The radiation is termed $K\alpha$ - or $K\beta$ -radiation, if the previous orbital was $2p$ or $3p$, respectively. Furthermore, one distinguishes $K\alpha_1$ - and $K\alpha_2$ -radiation, depending on the magnetic quantum number of the previous orbital, which can be $\frac{3}{2}$ or $\frac{1}{2}$, respectively. $K\alpha$ -radiation is desired for probing crystal structures.

Bibliography

- [1] Neil W. Ashcroft and N. David Mermin. *Solid State Physics*. Saunders College Publishing, 1976. 826 pp.
- [2] George F. Harrington and José Santiso. “Back-to-Basics tutorial: X-ray diffraction of thin films”. In: *Journal of Electroceramics* 47.4 (2021), pp. 141–163. ISSN: 1385-3449, 1573-8663. DOI: [10.1007/s10832-021-00263-6](https://doi.org/10.1007/s10832-021-00263-6).
- [3] Lothar Spieß, ed. *Moderne Röntgenbeugung: Röntgendiffraktometrie für Materialwissenschaftler, Physiker und Chemiker*. 2., überarb. und erw. Aufl. Studium. Wiesbaden: Vieweg + Teubner, 2009. 564 pp. ISBN: 978-3-8351-0166-1.