

# Machine Learning Applied to Materials Science

## Hypergraphs, Equivariant Networks, and Tensors

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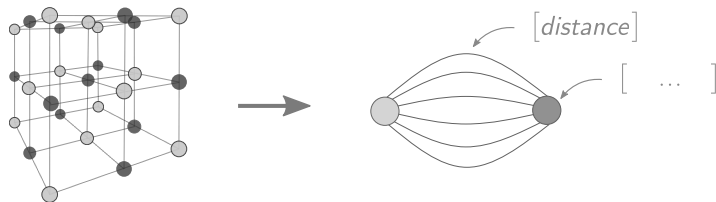


- Crystal graphs, MPNNs, and hypergraphs

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- Equivariant Networks and Tensors

# Crystal Graph Construction

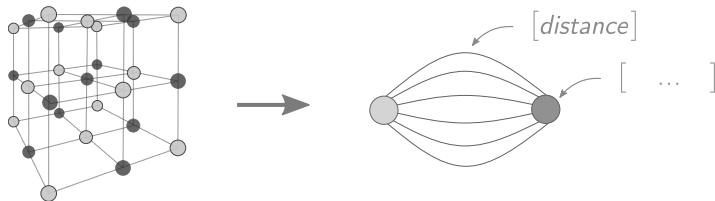
Usually represent crystalline systems as graphs (nodes and edges)<sup>1</sup>:



<sup>1</sup>Xie and Grossman, "Crystal graph convolutional neural networks for an accurate and interpretable prediction of material properties".

# Crystal Graph Construction

Usually represent crystalline systems as graphs (nodes and edges)<sup>1</sup>:



- Atoms are nodes
- Edges denote 'neighbors'; based on distance
- Feature vectors

<sup>1</sup>Xie and Grossman, "Crystal graph convolutional neural networks for an accurate and interpretable prediction of material properties".

# Message Passing on Graphs

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MPNN framework<sup>2</sup>:

Crystal Graphs

Crystal  
Hypergraphs

Results

Equivariance

NEquiP

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<sup>2</sup>Gilmer et al., “Neural message passing for quantum chemistry”. 🔍 ↻

# Message Passing on Graphs

MPNN framework<sup>2</sup>:


- Message  $m$  for each node  $i$ :

$$m_i^{t+1} = \sum_{n_j \in \mathcal{N}(i)} M_t(n_i^t, e_{ij}, n_j^t)$$

- Update function  $U$  for layer  $t$ :

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We'll need to generalize this later...

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# Graph Limitations

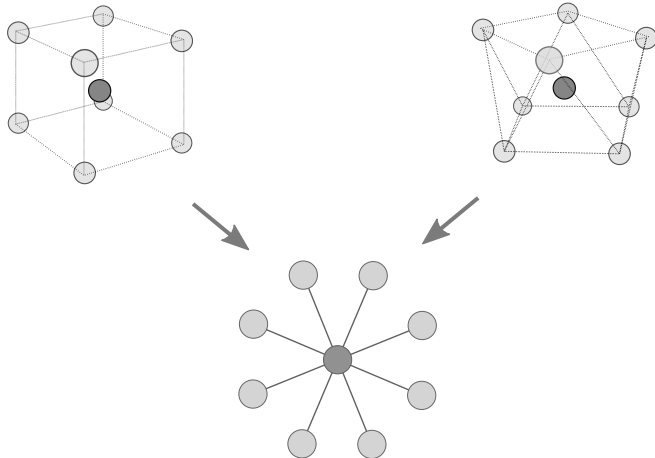
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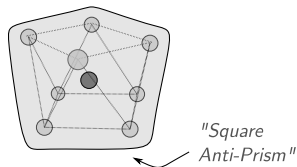
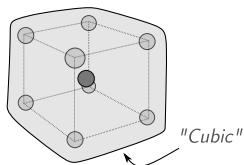
These two crystalline structures have same representations!

# Solution: Hypergraphs!

Hypergraphs allow us to have larger hyperedges.

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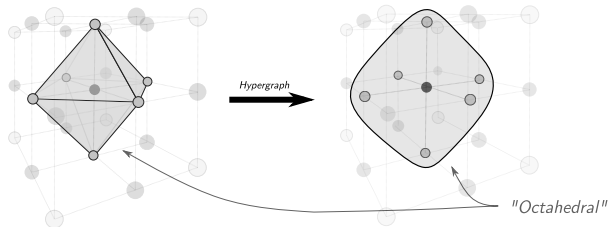
Hypergraphs allow us to have larger hyperedges.



So we may define hyperedges for this local geometry.

# Local Environments as Hyperedges

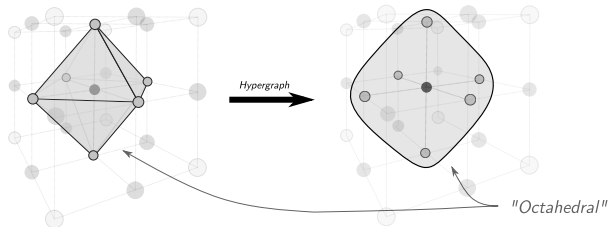
Construct a hyperedge for each atom's entire local environment:



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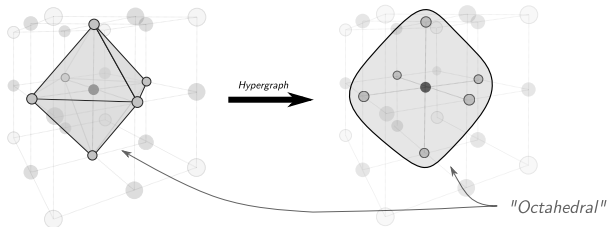


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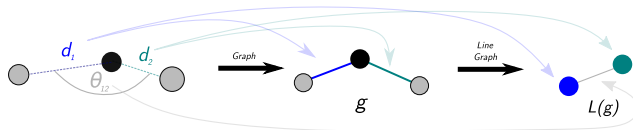
- Geometry is encoded quantitatively with continuous symmetry measures or structure order parameters<sup>3</sup> as features

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## Another Example: Triplets as Hyperedges

Many modern models utilize bond angle information<sup>4</sup>.

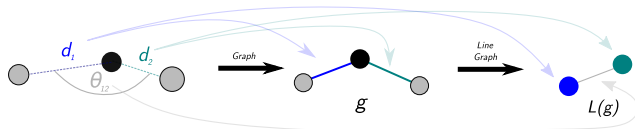


This is edge feature in a derived, auxiliary, 'line-graph'.

<sup>4</sup>Choudhary and DeCost, "Atomistic Line Graph Neural Network for improved materials property predictions"; Chen and Ong, "A universal graph deep learning interatomic potential for the periodic table".

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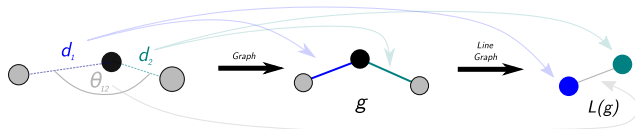
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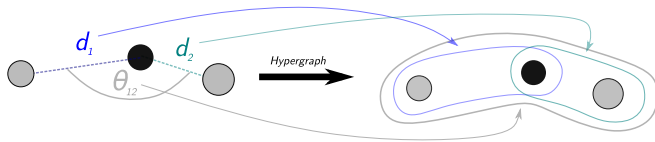
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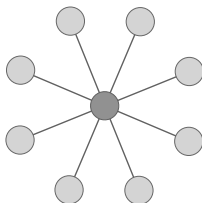
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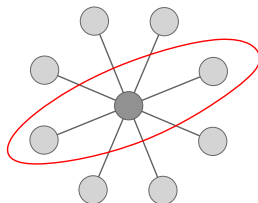
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# Crystal Hypergraphs



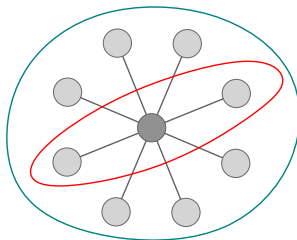
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# Crystal Hypergraphs



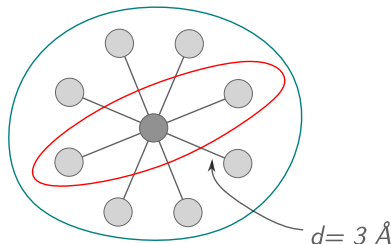
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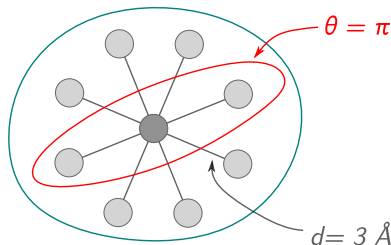
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- Each has hyperedge type with invariant features: distance

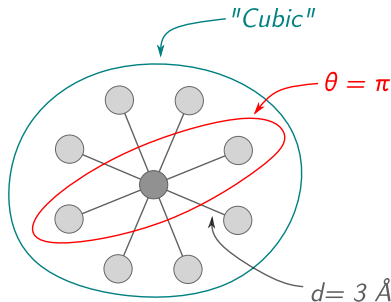
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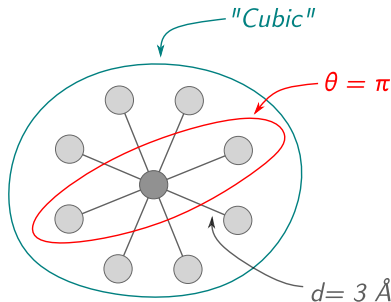


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But..... how do we update these representations?

# Extending Message Passing to Hypergraphs

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In the case of hypergraphs, neighborhood of nodes relevant to each message is now a set:

Crystal Graphs

Crystal  
Hypergraphs

Results

Equivariance

NEquiP

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$$m_i^{t+1} = \sum_{h_j \ni n_i} M_t(n_i^t, \underbrace{h_j^t, \{n_w^t | n_w \in h_j\}}_{e_{ij}, n_j})$$

$$n_i^{t+1} = U_t(n_i^t, m_i^{t+1})$$

# Comparative Performance Testing

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- We implement a basic convolutional structure amenable for (3)

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- We focus on models with bond hyperedges and motif hyperedges.
- Acts as testbed for what features/order correlations are most important for task.

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## Crystal Graphs

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For each convolutional structure, testing was done for a model with 3 convolutional layers, an initial learning rate of 0.01, hidden node features of dimension 64, and a hidden output layer of dimension 128 (Similar to CGCNN's architecture). The loss function utilized is MSE.

Datasets are split 80% for training and 20% for validation tests.





Below, we present results on validation sets for various MatBench target sets.

Model	Phonons (1,265) Best MAE (cm <sup>-1</sup> )	Refractive Indices (4,764) Best MAE
Bond-only	78.9	<b>.4891</b>
Motif-only	65.9	.5328
Bond & Motif	<b>59.0</b>	.5222

Model	Perovskites (18,829) Best MAE (eV/Atom)
Bond-only	0.329
Motif-only	<b>0.052</b>
Bond & Motif	0.058



- Electronic tasks (band gap, dielectric targets) seem to benefit much less, or be negatively impacted, by this additional information. It seems pair-wise correlators are, in general, a better descriptor for such tasks.

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## Overview of Experimental Results

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- Formation energy tasks benefit greatly from included motif (i.e. higher order geometrical) descriptors
- In the case of Perovskites (a harder set of formation energy targets), this geometrical information seems even more impactful. Though, it also may just help more for the smaller dataset available for training.







- Graphs are limited in their expression of higher-order (above pair-wise) features of crystal structures
- Crystal hypergraphs give us a natural way to encode different types of features in one representation

$$f(D_X(g) \cdot x) = D_Y(g) \cdot f(x)$$

Hence, equivariance conditions may often restrict our trainable parameter space (requiring less data in training) while providing similar expressibility of the model (giving similar, or sometimes better performance).

## Example: Convolutional Networks and Translation

Traditional convolutional neural networks (CNNs) are actually equivariant under the group action of translation (in 2D for the usual input of images).

The special orthogonal group of 3 dimensions,  $SO(3)$ , is the group describing 3 dimensional rotations.

It has irreducible representations indexed by a rotational order  $\ell$  and a harmonic order  $m$ , termed the spherical harmonics  $Y_m^\ell$ .

Note that products of spherical harmonics can be decomposed in terms of another linear superposition of spherical harmonics via Clebsch-Gordon Coefficients

$$C_{\ell_1 m_1 \ell_2 m_2}^{\ell_f m_f} Y_{\ell_1}^{m_1}(\Omega) Y_{\ell_2}^{m_2}(\Omega) = \sum_{\ell_3 m_3} \sqrt{\frac{(2\ell_1 + 1)(2\ell_2 + 1)}{4\pi(2\ell_3 + 1)}} C_{\ell_1 m_1 \ell_2 m_2}^{\ell_3 m_3} C_{\ell_1 0 \ell_2 0}^{\ell_3 0} Y_{\ell_3 m_3}(\Omega)$$

Layer-to-layer convolution defined as:

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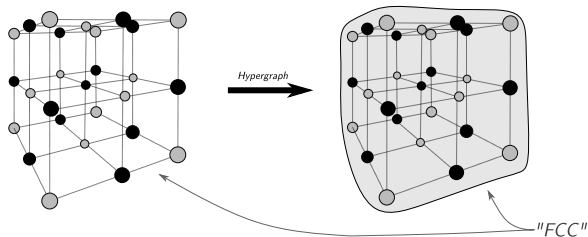
$$[[\text{node-index}, \dots],$$

$$[\text{hyperedge-index}, \dots]]$$



# Cell Hyperedges

Another order of hyperedge we may consider is that describing the entire unit cell of some crystalline structure.

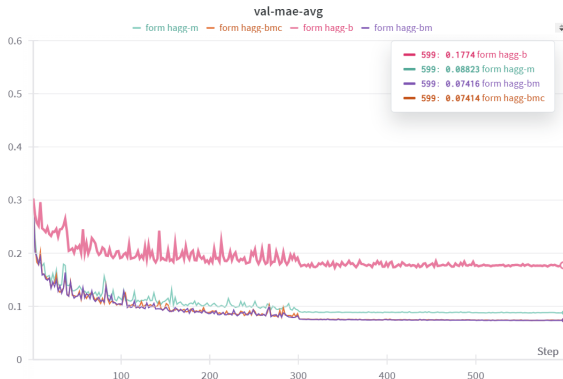


These 'unit' cell hyperedges allow for the explicit inclusion of global crystalline structure properties, i.e. point group information.

This unit cell feature also may be learned through convolution and potentially used as a dynamic state vector.

# (Materials-Project) Formation Energy

Below, validation MAE is shown through training for formation energy. The total dataset includes 152,605 materials.



Model	Best MAE (eV/Atom)
Bond-only	0.177
Motif-only	0.088
Bond & Motif	<b>0.074</b>

# (Materials-Project) Band Gap

Below are results for band gap training, based on a dataset including 152,605 materials.

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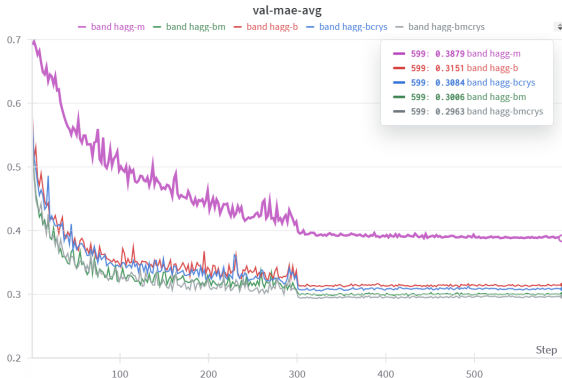
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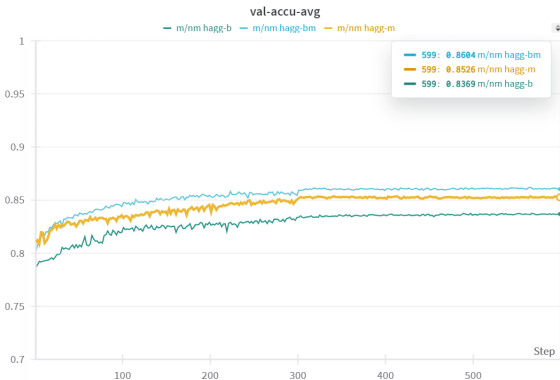
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Model	Best MAE (eV)
Bond-only	0.315
Motif-only	0.387
Bond & Motif	<b>0.301</b>

# (Materials-Project) Metal/Non-metal

Below, we test metal/non-metal classification for 152,605 materials.



Model	Best Accuracy
Bond-only	.837
Motif-only	.853
Bond & Motif	<b>.860</b>

# (MatBench) Perovskites

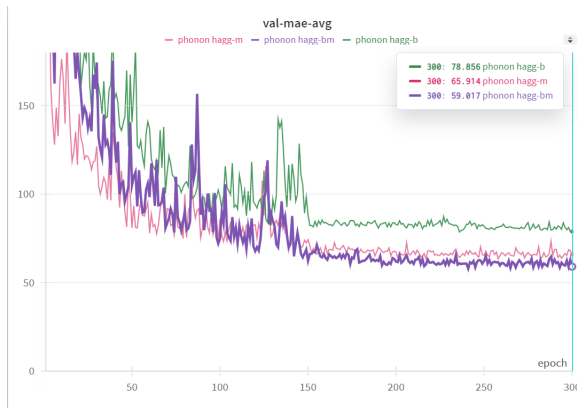
Below, we test on 18,928 calculated formation energies for Perovskites.



Model	Best MAE (eV/Atom)
Bond-only	0.329
Motif-only	<b>0.052</b>
Bond & Motif	0.058

# (MatBench) Phonons

Below, we test on a dataset of 1,265 materials with the target being the highest calculated frequency optical phonon mode peak.



Model	Best MAE ( $\text{cm}^{-1}$ )
Bond-only	78.9
Motif-only	65.9
Bond & Motif	<b>59.0</b>

# (MatBench) Dielectrics

Below, we test on the calculated refractive index of 4,764 materials

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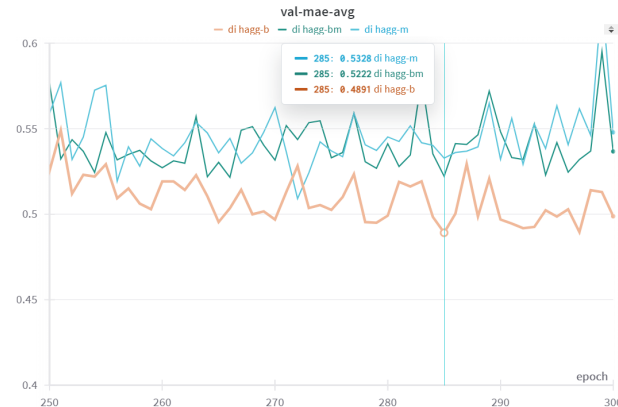
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