

$$I_{theoretical}(\vec{H}) \approx |F(\vec{H})|^2 \frac{V_{cry}}{V_{EC}^2} \underbrace{K(2\theta)}_{correction\ factor}$$

we want to compare the theoretical and the experimental intensities.

It's required to introduce a scale factor S related to the experimental set-up (that accounts for the crystal size, incident intensity, detector efficiency and more):

$$|F(\vec{H})|^2 \frac{V_{cry}}{V_{EC}^2} \underbrace{K(2\theta)}_{correction\ factor} \quad Scale \leftrightarrow I_{exp}(\vec{H})$$

we thus define $\Omega = \sum_{\vec{H}} \frac{[I_{theo}(\vec{H}) - I_{exp}(\vec{H})]^2}{\sigma^2(\vec{H})}$ in order to minimise this function.

$$F(\vec{H}) \approx \sum_{j=1}^{N_{atoms}} e^{2\pi i \vec{H} \cdot \vec{R}_{j,0}} e^{-B_j \vec{H}^2} f_j(\vec{H})$$

we start from the condition that we are approximating the coordinates of the atoms $\vec{R}_{j,0}$, then we will refine them. Same for B_j and S . This procedure is called STRUCTURE REFINEMENT.

B_j is known as thermal parameter or as "atomic displacement parameter".

STRUCTURE SOLVING

electron density is lattice invariant:

$$\rho(\vec{x}) = \rho(\vec{x} + \vec{T}) = \frac{1}{V} \sum_{\vec{H}} A(\vec{H}) e^{-2\pi i \vec{H} \cdot \vec{x}}$$

where $\vec{T} = m\vec{a} + n\vec{b} + p\vec{c} \in \mathbb{N}$ and where the summation over \vec{H} means the sum of hkl from $+\infty$ to $-\infty$ and $A(\vec{H})$ is a series of Fourier coefficients.

The Fourier coefficients correspond to the structure factor:

$$F(\vec{H}) \rightarrow A(\vec{H}) = \int_{V_{EC}} e^{2\pi i \vec{H} \cdot \vec{x}} \rho(\vec{x}) dV$$

experimentally is possible to notice that:

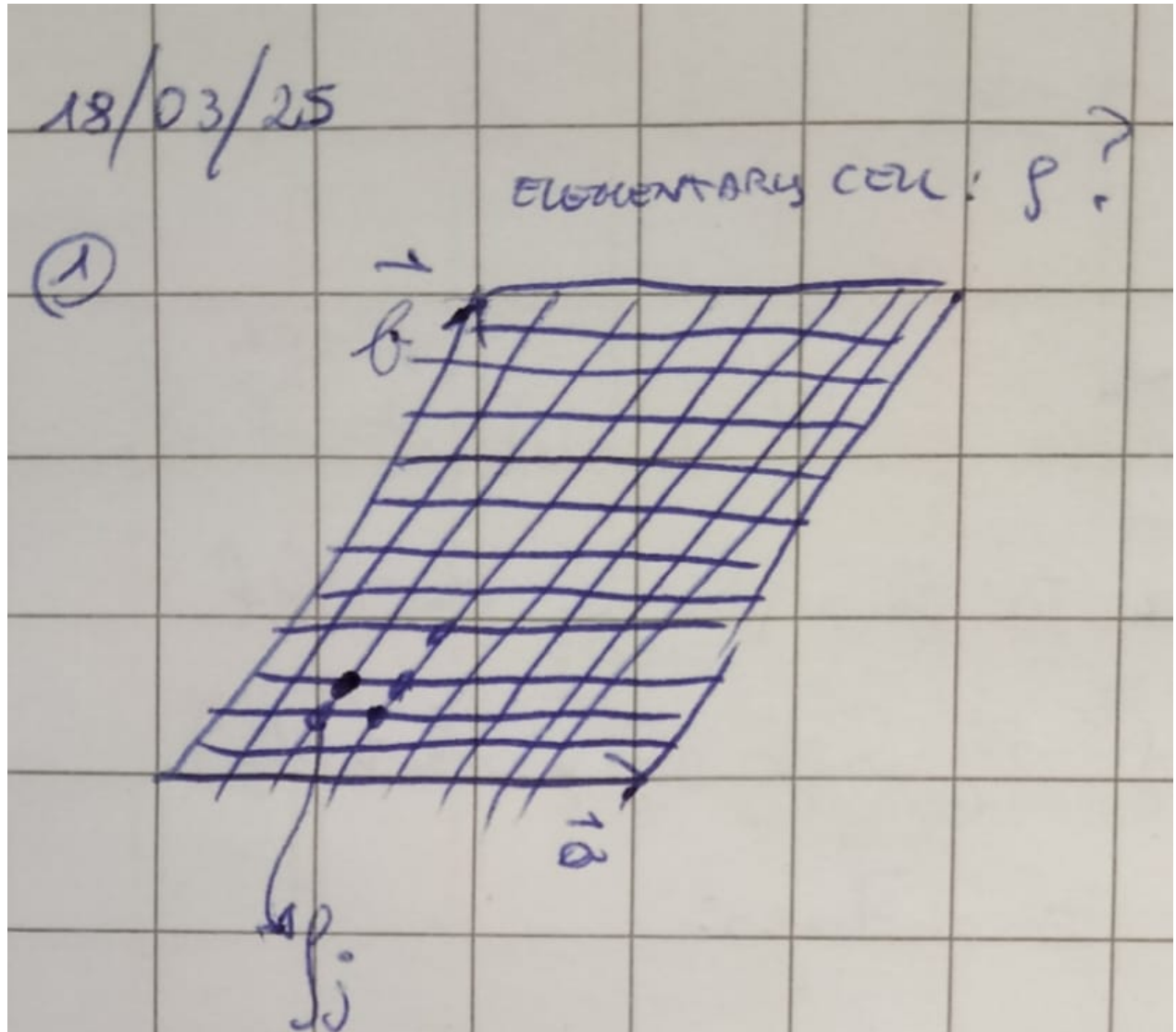
$$I_{observed}(\vec{H}) \propto |F(\vec{H})|^2$$

so, just because the structure factor is a complex function, we can just know its modulus.

$F(\vec{H}) = |F(\vec{H})| e^{i\phi(\vec{H})}$ ----> that's why the problem of defining the structure factor is called "the phase problem".

There are 3 methods used for structure solving:

1. maximum entropy method (MEM) we just want to find the electron density in a grid of pixels inside the elementary cell (see picture 1)



and we sink the electron density at each node (in the picture we see the j^{th} pixel). We don't know nothing about ρ_j , but we assume that ρ_j must be the best possible values that maximise the function

$$\Omega = \sum_{j=1}^{pixels} \rho_j \ln(\rho_j)$$

which is called "ENTROPY". So searching the $max(\Omega)$ (with the Lagrange multipliers method), we find the electron density.

When we have a set of $\{\rho_j\}$, we are able to calculate $F_{theo}(\vec{H})$. We also know that

$$\sum_{j=1} \rho_j = n \text{ electrons}$$

and we have the constraint that $\{\rho_j\} \rightarrow F_{theo}(\vec{H})$ where $|F_{theo}(\vec{H})| = |F_{exp}(\vec{H})|$

3) charge flipping method: tricky but effective

It exist a set of structure factors written as $F(\vec{H}) = |F_{exp}(\vec{H})| e^{i\phi_{guess}(\vec{H})} \rightarrow \rho_j^{[0]}$ where the apex $^{[0]}$ is the zero level of approximation and it's preferable to pixel the cell;

so we use the guess phase for the electron density: $\rho(\vec{x}) = \frac{1}{V} \sum_{\vec{H}} F(\vec{H}) e^{-2\pi i \vec{H} \cdot \vec{x}}$ and we take $|\rho_j^{[0]}|$, meaning that we change the negative sign in positive, when it occurs (we thus flip the charge). We thus obtain $\rho_j^{[1]}$. Now we calculate the structure factor associated with this electron density: $F^{[1]}(\vec{H})$. For each structure factor, I can determine the phases $\phi^{[1]}(\vec{H})$. Taking these phases, I reconstruct a new structure factor, that is $|F_{exp}(\vec{H})| e^{i\phi^{[1]}(\vec{H})}$. I repeat the procedure as an iterative process.

4. direct methods (set of methods with common basic idea): we know how to solve

$$F(\vec{H}) = \underbrace{|F(\vec{H})|}_{\text{experimental}} * \underbrace{e^{i\phi}}_{\text{missing}}$$

we will not see in detail, but there are techniques able to investigate the probability of having a given phase ϕ .

- $$f_j(\vec{H}) = \int e^{2\pi i \vec{H} \cdot \vec{x}} \rho_{j,atom}(\vec{x}) dV \approx \int e^{2\pi i \vec{H} \cdot \vec{x}} \rho(|\vec{x}|) dV = f_j(|\vec{H}|)$$

by the way, using the Bragg's law, we have that $|\vec{H}| = \frac{2}{\lambda} \sin\theta$. We find in the literature a dependence $f_j(\frac{\sin\theta}{\lambda})$ instead of $f_j(|\vec{H}|)$.

- $\rho_{j,atoms}(\vec{x})$ can be modelled (with LDA, DFT, HF ...) and minimized with the refinement but the position j^{th} has to be known

Parametrisation of $f_j(|\vec{H}|)$:

$$f_j\left(\frac{\sin\theta}{\lambda}\right) = c_0 + \sum_{k=1}^{4,6} a_k e^{-\left(\frac{\sin\theta}{\lambda}\right)^2 b_k}$$

where the summation goes to 4,5,6 because it's enough to have a good estimation. In order to have a good precision, we can stop to 6.

ANOMALOUS SCATTERING

Larmor law: $\mathcal{P} \propto q^2 \vec{a}^2$

The acceleration, $\vec{a} = \ddot{\vec{x}} = \frac{\vec{E}q}{m}$ can be expressed introducing a term of velocity,

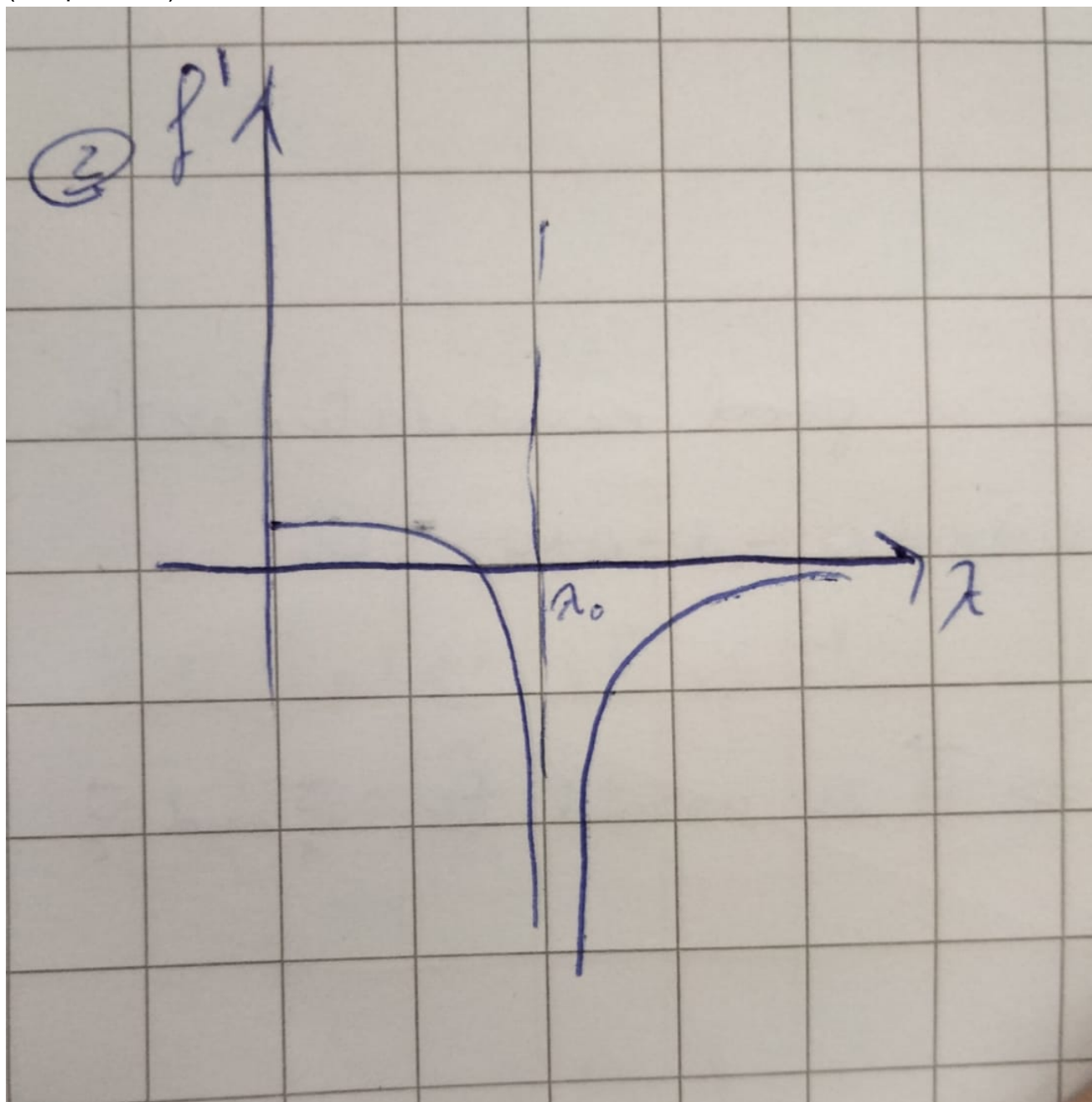
$$\ddot{\vec{x}} + \underbrace{\gamma \dot{\vec{x}}}_{\text{damping factor}} = \frac{\vec{E}q}{m}$$

by introducing the damping factor, we cannot describe $f(\vec{H})$ as we said before, but we have to introduce a couple of terms more:

$$f(\vec{H}) = \underbrace{\int e^{2\pi i \vec{H} \cdot \vec{x}} \rho_{atom}(\vec{x}) dV}_{f_0(\vec{H})} + \underbrace{f'(\lambda) + i f''(\lambda)}_{\text{anomalous contribution}}$$

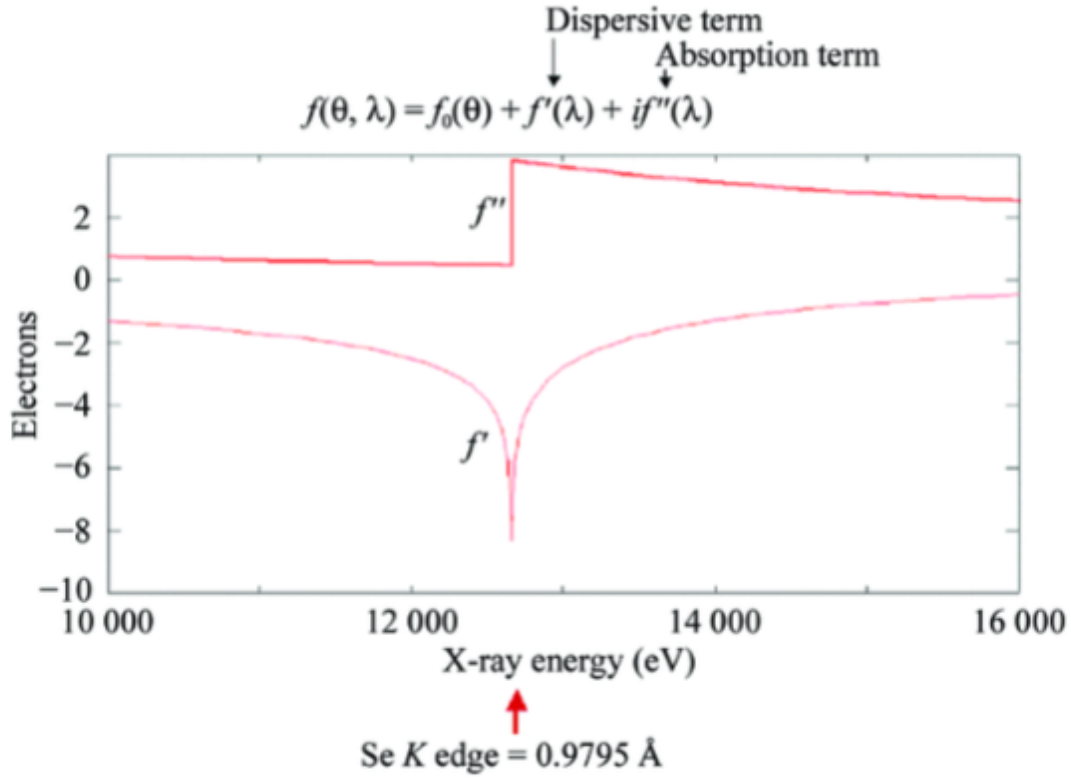
the anomalous contribution is relevant only in resonant conditions.

(see picture 2)



where λ_0 is the resonance wavelength, at which f' becomes extremely important. λ_0 depends on the chemical species.

By looking at f'' , we see a step at the resonance wavelength (or resonance energy):



HOW TO TREAT THE CASE OF SOLID SOLUTION

$$F(\vec{H}) = \sum_{j=1}^{atoms} e^{2\pi i \vec{H} \cdot \vec{x}_{j,0}} * e^{B_j \vec{H}^2} \underbrace{\left[f_{0j}\left(\frac{\sin\theta}{\lambda}\right) + f'_j(\lambda) + if''_j(\lambda) \right]}_{\text{anomalous scattering } f_j(\frac{\sin\theta}{\lambda}, \lambda)}$$

and we assume that there are no vacancies, so the sum of each probability of occupancy in each site j is 1: $P_j^{Mg} + P_j^{Fe} = 1$ for instance. The contribution of occupancy of each site is represented generally by

$$e^{2\pi i \vec{H} \cdot \vec{x}_{j0}} e^{-B_j \vec{H}^2} \left[P_j^{Mg} f_{Mg}\left(\frac{\sin\theta}{\lambda}, \lambda\right) + P_j^{Fe} f_{Fe}\left(\frac{\sin\theta}{\lambda}, \lambda\right) \right]$$