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MECHANICAL PROPERTIES OF METALLIC COMPONENTS ADDITIVELY MANUFACTURED BY POWDER FED LASER BASED DIRECTED ENERGY DEPOSITION

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SUMÁRIO

[1 INTRODUCTION 3](#_Toc21668567)

[1.1 GOALS 6](#_Toc21668568)

[1.1.1 General goals 6](#_Toc21668569)

[1.1.2 Specific goals 7](#_Toc21668570)

[2 BIBLIOGRAFIC REVIEW 7](#_Toc21668571)

[2.1 LASER 7](#_Toc21668572)

[2.2 ADDITIVE MANUFACTURING 9](#_Toc21668573)

[2.2.1 Directed Energy Deposition 11](#_Toc21668574)

[2.2.2 Powder fed Directed Energy Deposition Laser (PL-DED) 11](#_Toc21668575)

[2.2.3 Process description 13](#_Toc21668576)

[2.2.4 Thermal cycles and resulting properties 15](#_Toc21668577)

[2.2.5 Influence of residual porosity in mechanical properties 16](#_Toc21668578)

[2.2.6 Reported mechanical properties and sampling strategy 17](#_Toc21668579)

[3 MATERIAIS E MÉTODOS 19](#_Toc21668580)

[3.1 METODOLOGIA DE ENSAIOS E DESENVOLVIMENTO 19](#_Toc21668581)

[3.2 RECURSOS TÉCNICOS 19](#_Toc21668582)

[2.1.1. Sistema laser 19](#_Toc21668583)

[3. CRONOGRAMA DE TRABALHO 20](#_Toc21668584)

[4 BIBLIOGRAFIA 20](#_Toc21668585)

# INTRODUCTION

Additive manufacturing (AM) has gained attention for the last two decades for its great potential to reduce the time between ideation and production of complex geometries. This characteristic is being frequently explored among researchers and engineers to evaluate and compare product concepts beyond virtual models and prior to investing in tooling for mass production. The process reduced supply chain, also has an enormous potential to reduce the lead time between a sales order and the delivery of a functional component, consequently decreasing inventory cost [1].

Another great potential of the technology is the reduction of geometric constraints, removing barriers to the production of components optimized for multi-physics restrictions. Such optimizations can result in reduction of assembly part count and component weight, leading to more efficient operation and less fuel consumption in systems such as airplanes [2][3]. Figure 1 illustrates applications of different AM technologies used in the aerospace industry.

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| (a) | (b) | (c) | (d) |

Figure 1 - Examples of additive manufacturing of metallic components. (a) Titanium bracket connector produced on board the Airbus A350 XWB; (b) Turbine housing fabricated by the LASERTEC 65 3D System using multi-axis deposition; (c) Exhaust duct fabricated using the Laser Engineered Net Shape (LENS) process [2];(d) Repairing for damaged titanium blisk [4].

Despite the promising features AM is still expensive when compared to other manufacturing processes, especially for processing metals, with only a few technologies applicable to manufacture dense components. To accomplish that, Powder Bed Fusion (PBF), Sheet Lamination (SL) and Directed Energy Deposition (DED) are some of the AM groups of process available, each one presenting particular strengths and limitations [1][5][6].

Among the above mentioned categories, Directed Energy Deposition Laser with Powder as feedstock material (DED-LP), a sub category of DED processes, offers advantages such as suitability to repair failed components [4], suitability to produce larger parts with higher build rates when compared to Powder Bed Fusion (PBF) technology [16] and the potential to manufacture components with variations in alloy composition along the build volume, an advanced class of materials known as Functionally Graded Materials (FGM) [1][7][8].

Although some authors argue that the production cost will drop as technology evolves, manufacturing metal components by AM is still only justifiable when a significant reduction in lead time, inventory cost, part weight or part count takes place [9]. Furthermore, the lack of diffuse knowledge of process dependent design guidelines and output material properties, restrict the choice of manufacturing process according to the availability of material data still in the design phase [10].

In this context, the mechanical properties resulting from the process, are of special interest given the fact that metals are extensively used to withstand the mechanical loads of a variety of applications. Information extracted from stress-strain curves such as Young’s modulus (E), elastic limit (Sy) and tensile strength (Su) are the baseline for the mechanical design of structural components as illustrates Figure 2.

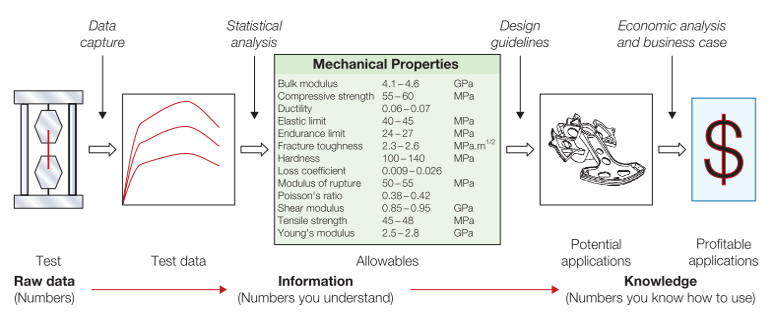


Figure 2 - Types of material information. Structured data for design “allowables” and the characteristics of a material that relate to its ability to be formed, joined, and finished; records of experience with its use; and design guidelines for its use [10].

Even though an enormous effort has been made for the last decade in order to standardize manufacturing practices and data reporting on mechanical properties, a collective understanding of the AM design paradigm as well as a database of material properties for different processing conditions is still required in order to democratize the technology [11][9].

## GOALS

### General goals

Considering the context described previously, the general goal of the present study is to propose a method to map the effects of key geometrical and processing parameters on the mechanical properties of samples produced by DED-PL. It is of the author’s interest, that the study finds its use as input information for product development engineers interested on prototyping or manufacturing functional components using the technology.

### Specific goals

In order to reach the general goal, the specific goals of the study are:

1. Select from the literature, critical geometric features that can induce variation in mechanical properties (e.g. aspect ratio, wall thickness, curvature radius and part height).
2. Select from the literature, critical processing parameters (e.g. processing speed, laser power and overlap distance) that can induce mechanical property variation;
3. Define processing parameter levels that result in a stable process for manufacturing the selected geometric features;
4. Successfully manufacture geometries with selected features and machine local samples for uniaxial tensile tests:
   1. Aligned to build direction Z;
   2. Aligned with perpendicular direction X;
5. Conduct hardness, microhardness and tensile tests to characterize:
   1. HB hardness;
   2. HV microhardness along the build direction Z;
   3. Elastic modulus (E);
   4. Elastic limit (Se);
   5. Tensile strength (Su);
   6. Elongation at fracture (ε);
6. Report the dependency of the measured values on the build direction and other geometric and process parameters;

# BIBLIOGRAFIC REVIEW

## LASER

The term “laser” was originally an acronym for *Light Amplification by Stimulated Emission of Radiation*, characterizing a special process of light amplification, but is often used to represent a special source of light. Laser, as a special type of electromagnetic radiation, can be described by its wavelength (λ) and intensity (I), differing from general light in its high degrees of directionality, monochromaticity and coherence [19].

The history of laser began with an appropriate description of electromagnetic radiation formulated by Maxwell in 1873. The following discoveries included the processes of *spontaneous emission* and *absorption* and the process of *stimulated emission* postulated by Einstein in 1917. In 1960, Theodore Mainman reported about the pulsed laser activity of a ruby laser for the first time [20].

Table 1 - Some important commercial lasers. Adapted from [19].

|  |  |  |
| --- | --- | --- |
| **Laser** | **Wavelength** | **Average power range** |
| Carbon dioxide (CO2) | 10,6 µm | Milliwatts to tens of kilowatts |
| Nd:YAG | 1,06 µm | Milliwatts to hundreds of watts |
|  | 532 nm | Milliwatts to watts |
| Nd:glass | 1,05 µm | Watts |
| Diodes | Visible and IR | Milliwatts to kilowatts |
| Argon-ion | 514,5 nm | Milliwatts to tens of watts |
|  | 448,0 nm | Milliwatts to watts |
| Fiber | IR | Watts to kilowatts |
| Excimer | Ultraviolet | Watts to hundreds of watts |

Since 1960, different types of laser have been developed as listed in Table 1, greatly varying in terms of dimensions, output power and emission wavelength (λ) [20], the last characteristic being of special importance to understand the interaction between laser and matter.

According to Poprawe [21] the fundamental principles involved in laser material processes are related to the absorption of light by the workpiece and its partial conversion to heat. The absorption depends on the wavelength and polarization of the laser light and also on the material physical properties as well as the characteristics and geometry of the workpiece surface [21]. Figure 4 illustrates absorption coefficients of different metals as a function of the wavelength of the incident radiation and shows that certain laser sources are better absorbed by specific materials.

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| (a) | (b) |

Figure 4 – (a) Absorption coefficient of a variety of metals as function of the wavelength of the incident radiation [22]; (b) Example of focusing feature of disk laser beam and power density distribution at focal point [23].

The laser energy is not homogeneously distributed along a cross section of a laser beam and its distribution depends on the wavelength, beam quality and the focusing optics. From the irradiated surface, a heat front moves into the inner material by conduction due to the developed temperature gradients. The heat flux lead to rise in temperature, depending on the absorbed intensity, the duration of the interaction, the beam radius at the surface, the velocity of the workpiece relative to the laser beam and the thermo-physical parameters of the material like heat conductivity and heat capacity [21].

High energy intensities available at a small controllable area on the intersection between the laser beam and the workpiece surface, make the laser a well-suited tool for additive manufacturing processes. With the right combination of laser source and material, it is possible to melt and deposit metal on specific regions of a workpiece creating metallurgical bonds, layer by layer, until whole functional components are built.

## ADDITIVE MANUFACTURING

Additive manufacturing (AM) can be defined as a process of joining materials to make parts from 3D model data, usually layer upon layer, as opposed to subtractive manufacturing and formative manufacturing methodologies [6].

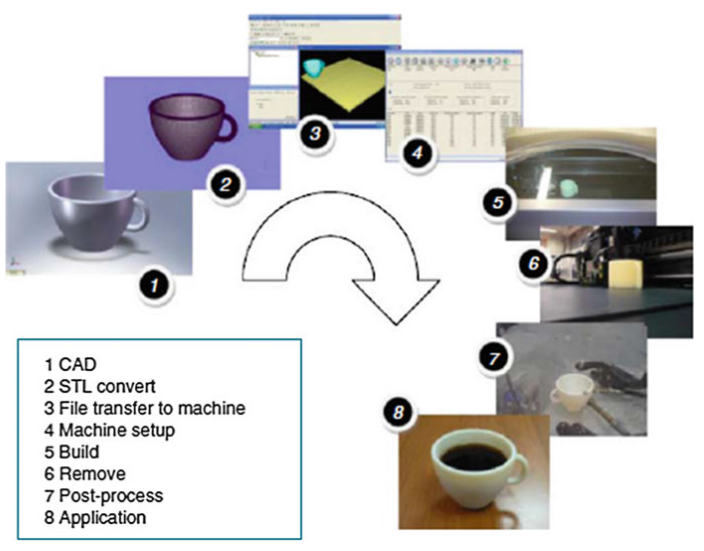


Figure 5 - Generic process of CAD to part, showing eight stages [24].

The idea was first documented in 1892 when Blanther applied the technique for building molds for topographic maps in three dimensions, but now it has developed into a highly digital manufacturing process, usually following the steps illustrated in Figure 5 to convert a digital 3D geometry into a physical part. The starting point is a 3D CAD model that is virtually sliced into thin layers with layer thickness of 20 µm – 1 mm, depending on the AM process. Based on this data the physical part is then built by repetitive deposition of single layers and locally melting the material by a heat source [25]. The part is then removed and usually post-processed before being suitable for the application.

AM methods can essentially be classified by the nature and the aggregate state of the feedstock as well as by the binding mechanism between the joined layers of the material [25]. A complete list of categories is available in Table 2, highlighting the suitability of each category to produce dense metal parts.

Table 2 - Categories of additive manufacturing and suitability for producing dense metal parts. Adapted from [6], [5], [26].

|  |  |  |  |
| --- | --- | --- | --- |
| **Category** | **Example** | **Processed materials** | **Density of metal parts** |
| Binder Jetting | 3D printing; | Metals, Polymers, Composites, Ceramics | Low |
| Directed Energy Deposition | WAAM1, Laser Cladding; | Metals | High |
| Material Extrusion | FDM2 | Thermoplastics, Waxes | - |
| Material Jetting | Polyjet | UV curable resins | - |
| Powder Bed Fusion | SLS3, SLM4 | Metals, Thermoplastics, Ceramic | High |
| Sheet Lamination | Sheet forming | Metals, Polymers, Paper | Low |
| Vat Polymerization | Stereolithography | UV curable resins, Wax, Ceramic | - |
| 1 Wire Arc Additive Manufacturing;  2 Fused Deposition Modeling;  3 Selective Laser Sintering;  4 Selective Laser Melting; | | |  |

AM has attracted much attention over the past ten years due to its inherent advantages, such as high design freedom and short lead times [25] and although AM techniques were at first limited to rapid prototyping of porous structures, recent advancements in technology made possible to reliably manufacture even dense metal parts with certain AM processes, including steel, aluminum and titanium. In the present work, one of the processes suitable for processing metal (Directed Energy Deposition) will be discussed in further detail.

### Directed Energy Deposition

Directed Energy Deposition, or DED, is defined as an additive manufacturing process in which focused thermal energy is used to fuse materials by melting as they are being deposited [27]. DED systems comprise multiple categories of machines using laser beam (LB), electron beam (EB), or arc plasma energy sources as categorized in Figure 6. Feedstock material typically comprises either powder or wire. Deposition typically occurs either under inert gas (arc systems or laser) or in vacuum (EB systems) [27]. The process was a natural evolution of welding techniques and the first of its kind was LENS ®, documented in 1998 by the Sandia Labs in Albuquerque, New Mexico [1].



Figure 6 - Classification of Directed Energy Deposition (DED) systems [28].

### Powder fed Directed Energy Deposition Laser (PL-DED)

Among the DED systems illustrated in Figure 6, in the present work we focus our attention to the laser-based process (DED-L), using powder as feedstock material. The process is well suited for manufacturing finer details when compared to arc technologies yet without using expensive vacuum chambers such as the ones needed for electron beam systems [25]. These two factors combined, make the laser systems very attractive among other DED processes.

In contrast to powder bed technologies, DED-L technology provides a high build rate and allows for larger build volumes although presenting lower layer thickness resolution. Depending on the main parameters, build rates up to 300 cm3/h can be achieved using a layer thickness of 40 μm – 1 mm. Feed rates between 4 g/min and 30 g/min are realized for the deposition of metal powder [25].

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| --- | --- | --- |
|  | | |
| (a) | (b) | (c) |

Figure 7 – IN718 helicopter engine combustion chamber fabricated by a five-axis laser DED process. (a) Deposition showing the 2.5-dimensional tool path; (b) Multiaxis deposition; (c) Finished part [2].

Moreover, as the material is melted while being deposited, it is possible to use 4 or 5-axis systems, as show Figure 7, to build structures with high overhang angles, reducing or eliminating the undesired metal support structures and reducing component postprocessing time [29]. One particular benefit of this characteristic is the potential to deposit material onto curved surfaces and existing metal structures. For this reason, one of the DED technologies, referred to as laser Cladding, is often used to repair damaged parts, particularly for the aerospace industry [30].

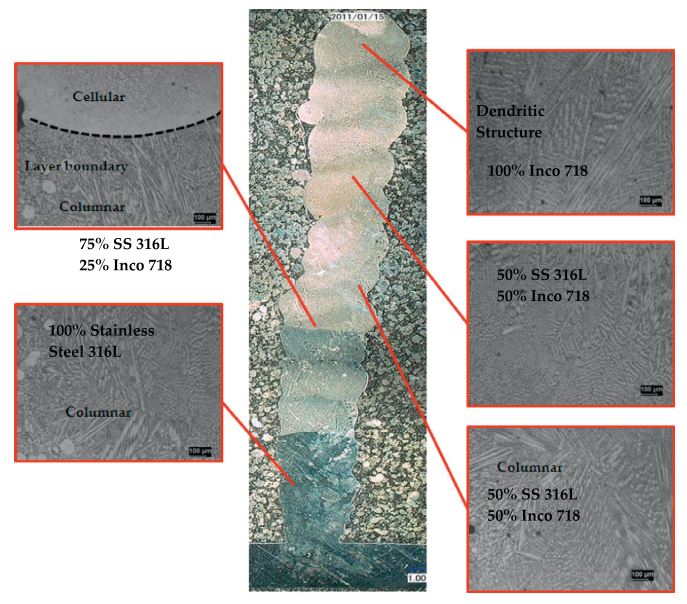


Figure 8 - Transverse cross section showing the transition between 100% 316L to 100% Inconel 718 from bottom to top [31].

Regarding the choice of feedstock material, one of the main advantages of using powder, is the capability of mixing different powders during the deposition, thus enabling the creation of gradient metal alloys in-situ, also known as functionally graded materials (FGM) as illustrates Figure 8. The concept extends the design space to components that can be made of a continuum structure that uses cost-intensive alloys in highly-loaded regions, and cheaper compositions in non-critical areas [32].

In order to make use of the full potentials of powder fed laser based DED process, a list of challenges and unanswered questions need to be addressed. This list includes uncertainties about predicting the geometry, metallurgical integrity and mechanical properties of the formed components. These outputs highly depend on interactions between laser beam, powder stream and scanning speed, which will be briefly discussed in the next sessions.

### Process description

A powder fed laser based DED system comprises five fundamental subcomponents: laser source, inert gas, powder feedstock feed mechanism, positioner and a computer control system. The process consists in the formation of a melt pool in the substrate through absorption and conversion of irradiated laser energy into heat. Simultaneously powder is fed into this pool and melts, subsequently solidifying to form a layer of deposited material as illustrated by Figure 9. A strong fusion bond between added material and substrate is achieved immediately [33].

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| --- | --- |
|  |  |
| (a) | (b) |

Figure 9 – (a) Lateral powder supply [33]; (b) Continuous coaxial powder nozzle during the deposition. (1) Process laser beam; (2) Focalizing lens; (3) Mix of metal powder with carrier gas; (4) Shielding inert gas; (5) Deposition nozzle; (6) Schematic (left) and realistic (right) views of a powder cone; (7) Melt pool; (8) Deposited structure; (9) Substrate. Adapted from [34] and [21].

A variety of nozzle configurations are reported in the literature, from lateral to coaxial, from discrete to continuous, two of which are schematically illustrated in Figure 9 (a) and (b). The powder is delivered to the melt pool by feeders, that can be based on various working principles, one of which consists of a container from which powder flows by gravity into a slot in a rotating disk. The powder is transported to a suction unit from which it is transported by a gas stream to a powder nozzle. The volumetric powder feed rate is controlled by the dimensions of the slot and the speed of the disk [33].

The nozzle head is programmed to move along a path, previously defined by the user, and as it moves, continuously deposit material on the surface of the substrate, usually maintaining a constant distance between the nozzle tip and the surface. After forming the first layer, the nozzle moves in the positive Z direction, represented in Figure 9, and starts to add material on top of the previously deposited layer. This process repeats until the entire part has been built. A schematic view of key input and output factors that affect the stability of the process are represented in Figure 9, some of which will be described in more detail later.

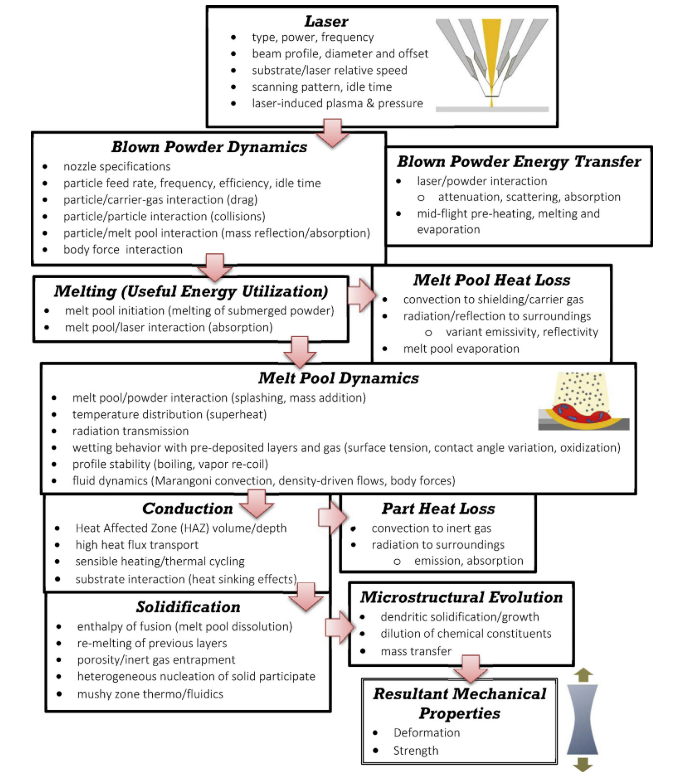


Figure 10 – Physical events occurring during DLD for given instant in time [35].

Along the process, several critical input variables schematically listed in Figure 10, interact with one another in a complex physical processes that involves light absorption, heat conduction and convection, mass diffusion and result in the geometry, microstructure, mechanical properties and other relevant characteristics [33]. The present study will focus on discussing geometric features, process parameters and their relationship with the resulting microstructure and mechanical properties.

### Thermal cycles and resulting properties

During AM, a defined volume element, or voxel, of the material is usually subjected to a complex thermal cycle. This thermal cycle involves a rapid heating above melting temperature due to the absorption of the laser energy and its transformation into heat, a rapid solidification of the molten material after the heat source has moved on, and numerous re-heating and re-cooling processes when the following layers are welded and the voxel is still exposed to heat [8]. The effect is better pictured in Figure 11 (a) derived from a numerical investigation conducted by Manvatkar et. Al [36].

AM microstructure is therefore a result of the described thermal cycle. Independently of the material, a fine-grained structure has usually been observed for AM in comparison to other processes (e.g. casting) [25]. The effect can be explained by the high cooling rates of up to 12000 K/s as shown in Figure 11 (b) when compared with 1 – 100 K/s for casting [7], which itself is a result of the very local heat input and the small volumes of molten material [25].

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| --- | --- | --- |
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| (a) | (b) | (c) |

Figure 11 – (a) Simulated thermal cycles at three monitoring locations in the first three layers in a DED-L of 316 stainless steel at a laser power of 210 W and 12.7 mm/s speed; (b) Variation of cooling rate at three monitoring locations in the three layers. The results of the heat conduction calculations are from the literature; (c) Computed and the experimentally determined hardness values in three layers compared with heat conduction model; [36]

The temperature gradients are obviously influenced by several process parameters, e.g. the energy density, laser power, scanning speed, layer thickness and pre-heating temperature, if applied [25]. Moreover, the temperature gradients are also affected by the surrounding material, and as heat conduction in build direction is typically higher than in other spatial directions as a result of the solidified material from lower, previously built layers, anisotropy in both microstructure and properties has been reported in several studies [25].

For constant laser power and scanning velocity, the layer width and peak temperature increase while the cooling rate decreases toward the top layers as shown in Figure 11 (b) extracted from simulated results [36]. As a result, the solidified deposit usually exhibits finer grain structure close from the substrate with coarser grains towards the top. Correspondingly, the yield strength and hardness also reduce from the bottom toward the top layers [37]. Figure 11 (c) illustrates simulated values that match with experimental data.

### Influence of residual porosity in mechanical properties

With advancements in AM technology over the past years, dense metallic parts with mechanical properties comparable to conventional manufacturing methods are achievable for several material and process combinations. As porosity facilitates crack propagation and deteriorates mechanical properties, the manufacture of parts with a high density, typically greater than 99.5%, is the first goal in AM process optimization [25].

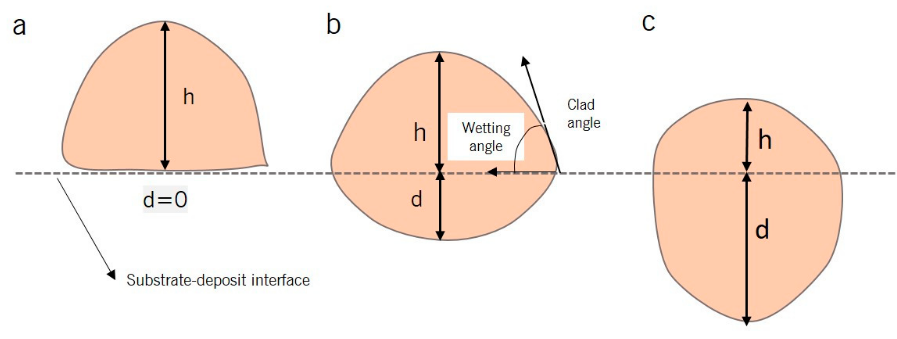


Figure 12 – (a) Zero or low penetration, meaning a lack of fusion with the substrate; (b) Optimal level of dilution (generally between 10% and 30%); (c) High d leads to keyholing [28].

Besides other influences, part density depends on the applied volume energy

|  |  |
| --- | --- |
|  | (1) |

Where is the laser power, stands for the scan speed, is the hatch distance and the layer thickness of the deposited layer. According to Herzog et. Al [25], too low energy input will result in unmolten material and thus reduced density by the formation of irregular-shaped voids. On the other hand, too high energy input will lead to higher melt pool dynamics and keyholing phenomenon resulting in spherical shaped pores formed due to entrapped gas, formed during evaporation of material, and reducing part density [25]. Figure 12 illustrates the schematics of weld pool cross sections of the above-mentioned circumstances and Figure 13 shows diagrams of the resulting pores geometry as well as a graph of the desired processing condition that generates minimum porosity.

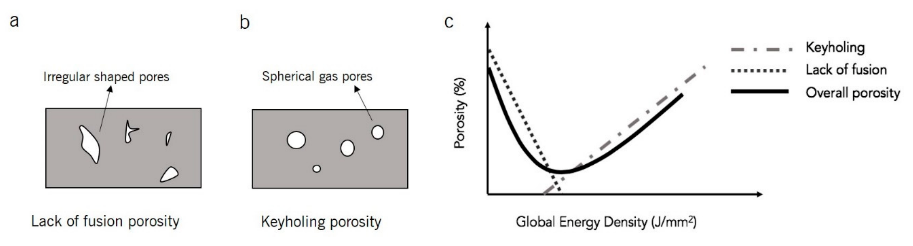


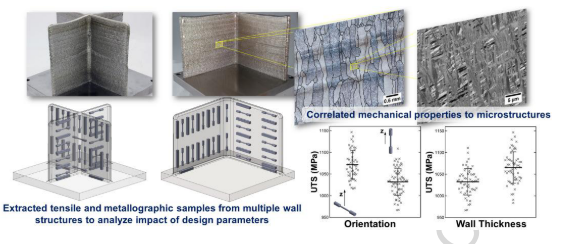
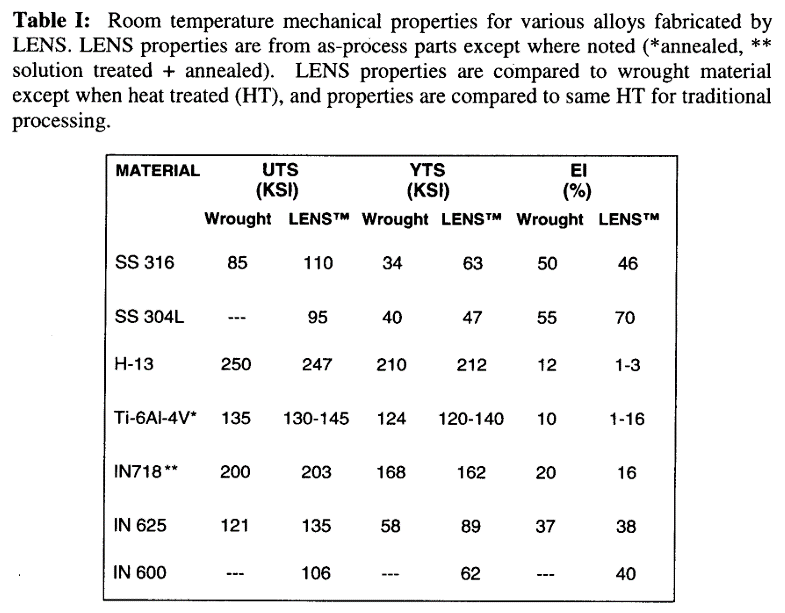
Figure 13 - Schematic showing: (a) Lack of fusion porosity (interlayer porosity), (b) keyholing porosity (intralayer porosity), and (c) the intersection of interlayer and intralayer porosity with respect volume energy [28].

### Reported mechanical properties and sampling strategy

The resulting mechanical properties of components manufactured by DED-L have been documented in the literature for a variety of materials and processing conditions with emphasis in those used in high end applications. The tensile strength (Su), yield strength (Sy), elongation (ε) and elastic modulus (E) of tool steels, high speed steels and high performance alloys such as Titanium and IN618 are more commonly found in the literature.

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| --- | --- |
|  |  |
| (a) | (b) |

Figure 14 – (a) Summary of Ti-6Al-4V AM tensile properties. Abbreviations: DMD, direct metal deposition; HT, heat treated; HIP, hot isostatic pressing; LENS, laser-engineered net shaping; UTS, ultimate tensile strength; YS, yield stress. Adapted from [38]. (b) Comparison of room temperature tensile properties for H13 tool steel shell build samples fabricated at various power and velocity values [39].



# MATERIAIS E MÉTODOS

## METODOLOGIA DE ENSAIOS E DESENVOLVIMENTO

O projeto inclui etapas para o desenvolvimento do sistema de alimentação dinâmica, integração do sistema ao microcontrolador e fonte de soldagem para o processo laser *cladding*. Os critérios de avaliação dos resultados alcançados ao longo do projeto serão baseados em dois aspectos principais. Um deles é a capacidade de integração entre a fonte de soldagem, microcontrolador e o próprio alimentador, o que implica em um levantamento de motores e drivers de potência compatíveis com as exigências do projeto.

Outro aspecto diz respeito à capacidade do sistema de executar sua função principal, ou seja, comprovar a integração mecânica, eletroeletrônica e computacional para a alimentação dinâmica e técnica de arame quente para o processo laser *cladding.* Para tal, o sistema será avaliado em relação a frequência de avanço e recuo do arame, assim como precisão e aumento de produtividade em relação ao processo convencional.

## RECURSOS TÉCNICOS

### Sistema laser

Os ensaios e construção do protótipo serão realizados no laboratório LMP. O laboratório possui um sistema de fonte LASER de fibra da IPS YLS-10000 com feixe de cerca de 300 mm no ponto focal e 880 de diâmetro. O cabeçote LASER utilizado consiste em duas partes: sistema ótico integrado ao sistema, sistema do gás de proteção como ilustrado na Figura 16.

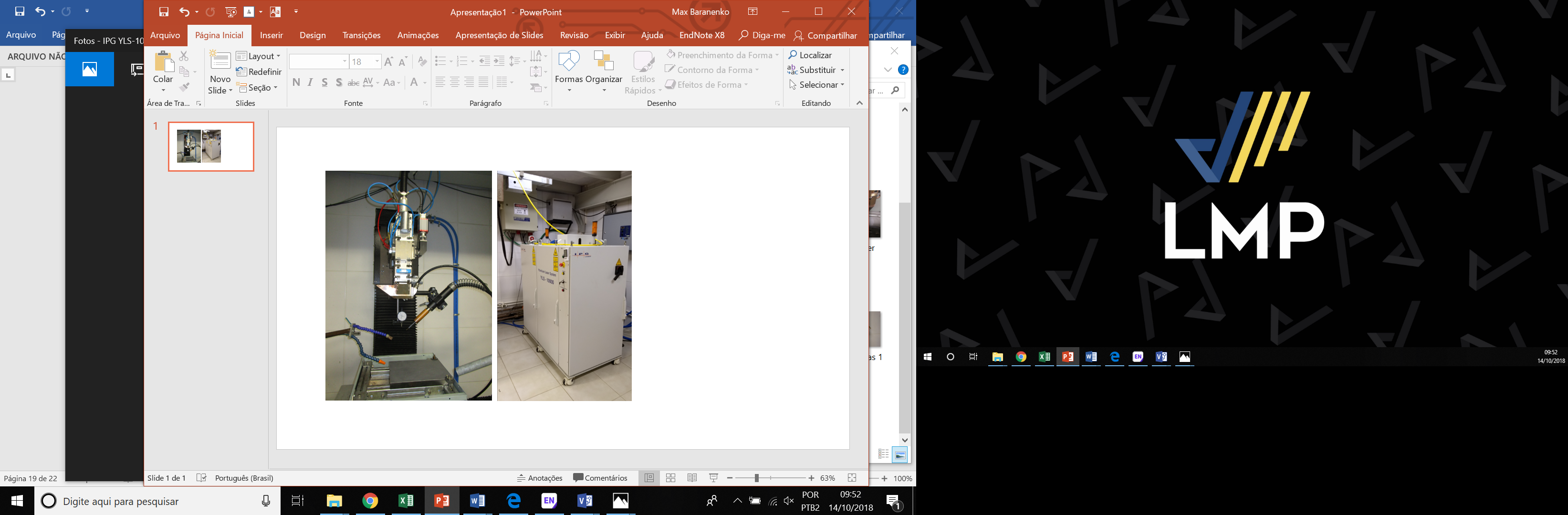
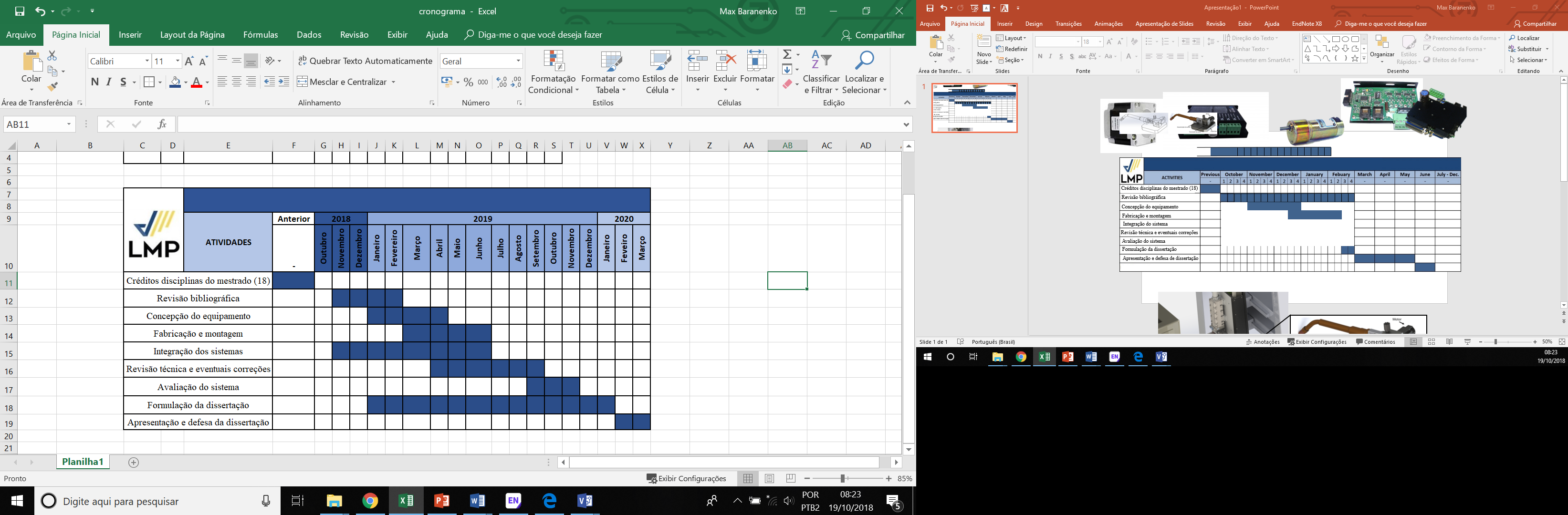


Figura 16– Esquerda Montagem cabeçote de soldagem para o processo laser *cladding;* Direita: fonte laser de fibra IPS YLS-10000 (Autoria própria).

# CRONOGRAMA DE TRABALHO

O cronograma a seguir (Quadro 1) ilustra a distribuição das principais etapas pertencentes ao projeto de dissertação de mestrado.

Quadro 1- Cronograma de atividades do projeto de dissertação. [35]



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