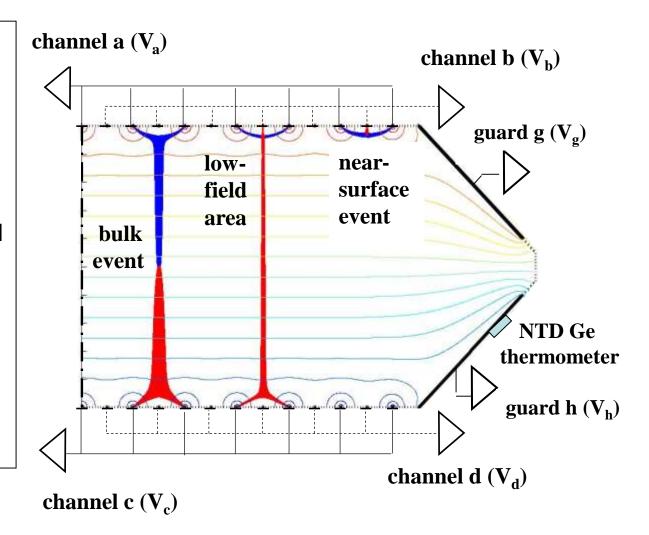
### PICTS of impurities and defects in Germanium

N. Fourches - CEA-Saclay/IRFU/SEDI EDELWEISS collaboration meeting

- 1 Thermal spectroscopy of impurities and defects
- 2 Incidence of electrically active defects on detector operation
- Dopant species, deep levels, surface states
- 3 Materials studies, detector validation and commissioning
- -"On-line" characterization of detector material
  - → PICTS, ODLTS, PTIS other?

## InterDigitized detectors:

Complex electrode configuration and biasing scheme compared with simple p+in+ structures: need to investigate material issues on both detector and material



\* Voltage biases:  $V_a = 2V$ ,  $V_b = 1V$ ,  $V_c = -2V$ ,  $V_d = -1V$ ,  $V_g = 0.5V$ ,  $V_h = -0.5V$  2

# 1- Impurities and defects in detector material: incidence on charge collection at cryogenic temperatures

#### **Issues:**

- Operating conditions to ensure optimal charge collection efficiency
- Trapping effects, space-charge build-up & detector reset

#### **Methods:**

- -Direct studies of the detector:
- -Charge collection efficiency measurements using radioactive sources & pulsed LEDs
- -Time-resolved acquisition of the charge collection signals -Material studies :
- -Thermal and optical spectroscopy using electrical signals for high sensitivity

- 2 Incidence on detector operation:
- a) Trap densities, orders of magnitude in HP-Ge

Trapping length: 
$$\Lambda_t = 1 / < \sigma N_t >$$

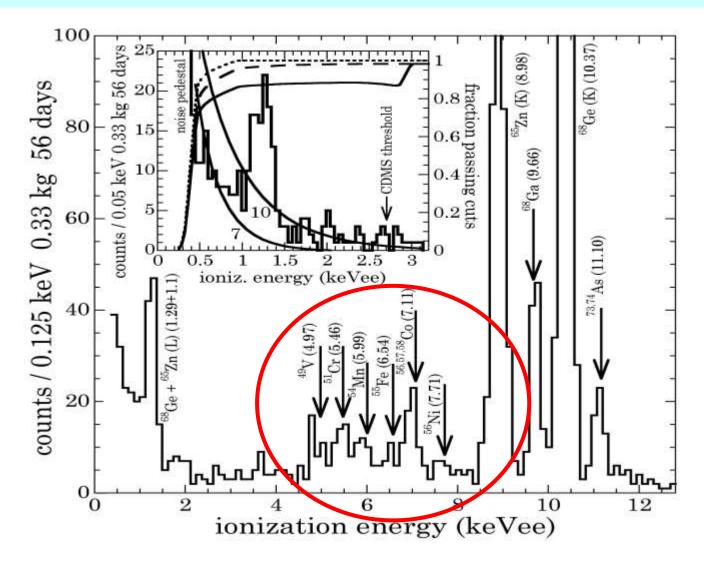
- (1)  $\Lambda_t$  is a function of the crystal quality (impurities, point defects, dislocations), of the charge state of the defects (ionized vs. neutral), and of the nature (electron or hole) and velocities of the carriers.
- (2) Orders of magnitude for hot electrons in HP-Ge at cryogenic temperatures (20 mK) (velocities of a few 10<sup>6</sup> cm/s): Depending on field strength,  $\Lambda_{\rm t}$  @ a few cm up to several tens of cm in a 'well regenerated' crystal.
- (3) A simple estimate:

Take  $N_t = 10^{10}$  cm<sup>-3</sup> (HP-Ge)  $\rightarrow \sigma$  has to be of the order of  $10^{-11}$  cm<sup>2</sup>, which would be expected for carrier trapping by ionized, rather than by neutral defects.

But this is clearly inconsistent with the assumption of a full neutralization of the charged defects as expected from the reset procedure by gamma or IR irradiation.

→ Either defect neutralization is less effective than we think, or there are more electrically active defects than we think!

#### b) More specific issues: transition metal impurities in HP-Ge



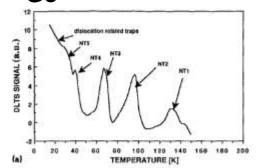
Low-energy spectrum after all cuts, prior to efficiency corrections. Arrows indicate expected energies for all viable cosmogenic peaks.

(C.E Aalseth ,Collar et al., arXiv:1002.4703v2 [astro-ph.CO] 2010).

# 4 - Materials studies, detector validation and commissioning

# Deep level characterization in ultra-pure germanium

#### N. Fourches et al., 1991 DLTS on ultra-pure Ge



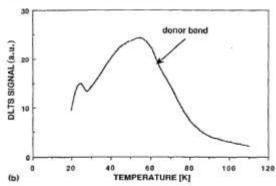
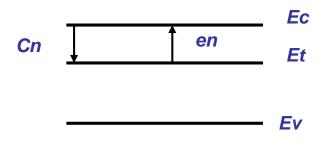
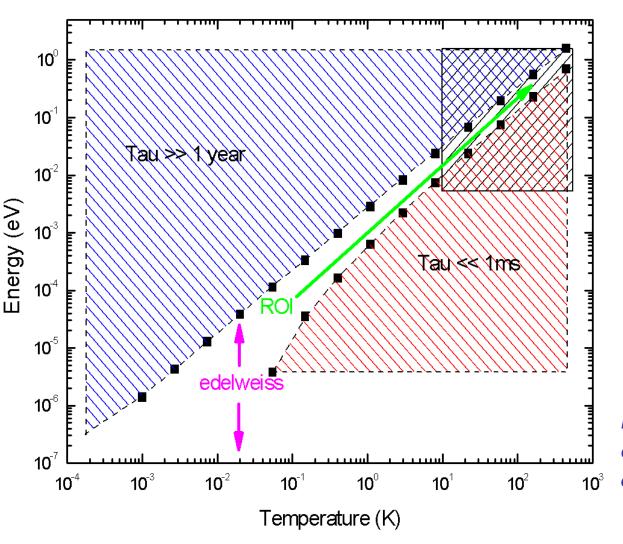


FIG. 2. (a) Typical DLTS spectrum obtained in p-type high-purity germanium: NT1-NT5 are the native deep levels (experimental conditions are emission rate: 56 s<sup>-1</sup>; filling pulse width: 1 ms; reverse bias: – 8 V; filling pulse bias: 0 V). (b) Typical DTLS spectrum obtained in As-grown n-type high-purity germanium showing the presence of a broad peak (experimental conditions are emission rate: 56 s<sup>-1</sup>; filling pulse width: 1 ms; reverse bias: – 8 V; filling pulse bias: – 1 V).

- How to perform deep level thermal spectroscopy directly on detectors?
- Use photo-induced current transients.
- Some advantages:
  - would apply to detector fully mounted,
     fitted with regeneration LED facing Ge crystal.
  - optical excitation allows to separate the electron and hole trap centers contributions by simple reversal of the electrode bias polarities.



# "standard" thermal spectroscopy: range of investigation



For a capture cross section: 10<sup>-15</sup> cm<sup>2</sup>
Emission rate >>detrapping time constant: Tau)

- Improvements required
- capture and emission behaviour in the temperature range of Edelweiss still under study

Emission rate =Nc<vth> $\sigma_n$ exp(-Ea/kT) derived from the detailled balance equation

## Photo Induced Current Transient Spectroscopy

```
Background : Capacitance Transient Spectroscopy of
Deep Levels in Semiconductors (D.V. Lang 1974)
PICTS: introduced by Hurtes et al. (1978), Merlet et al. (1978)
predominantly for high resistivity Semiconductors
(GaAs, InP ...)
Materials studied : Pbl<sub>2</sub> (Zielinger et al. 1985)
GaAs (Hurtes et al. 1978) Pbl<sub>2</sub>, Cu<sub>2</sub>O ( J.C. Balland ,1984),
CdTe (P. Wurm ,1995 )
CdTe (Zielinger et al. 1993, Castaldini et al., 1997)
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# Photo Induced Current Transient Spectroscopy

General principle (1978)

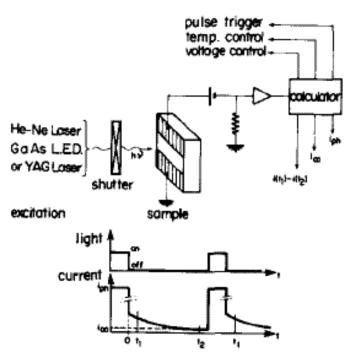


FIG. 1. Principle of the experiment.

- Simple experimental set-up
- No need for extensive signal processing
- High sensitivity
- Bulk or near surface defects and impurities interface states contribute but are not the primary interest of the technique
- Usable on a variety of samples and geometries Oxford 23-09-2010

## **Environment**

- New context : EURECA with many more detectors
- Cost: use of less pure material
- A post-process control would then be welcome. What is an appropriate technique?
- Probe the contamination of the material
- Non destructive
- High sensitivity: <<10<sup>9</sup> /cm<sup>3</sup>, DLTS and PICTS, OK
- Contactless: PICTS no, optical techniques & EPR yes
- Use of real detectors with their LEDs for optical injection. Current measurements: PICTS
- Usable in a final step when the germanium detectors are ready to be installed. No more handling then needed

# **Background**

- DLTS (1974, Lang), DLOS (1979, Chantre),TSC (1960),TSCAP (1972)
- Class of experimental techniques based on the relaxation of carriers trapped on defect (impurity) levels

Detection mode	Excitation mode	Stimulation mode	
Capacitance(Current)	Electrical	Thermal ( DLTS)	
Capacitance	Thermal	Thermal (TSCAP)	
Capacitance	Optical	Thermal (ODLTS)	
(Trans)conductance	Electrical	Thermal ( C-DLTS)	
Current	Thermal	Thermal (TSC)	
Capacitance	Electrical-Optical-Thermal	Optical (DLOS)	
Current	Electrical-Optical-Thermal	Optical (DLOS)	
Current	Optical	Thermal (PICTS)	

# Background

- Physical quantities used to build the signal (DLTS, PICTS...) are simply analyzed through a time constant filtering
- The experiment consists in fact in an energy scan using the temperature (DLTS) or the wavelength (DLOS)
- Numeric DLTS is now commonly implemented using a sampling and digitizing board which allows the digitization of the transient

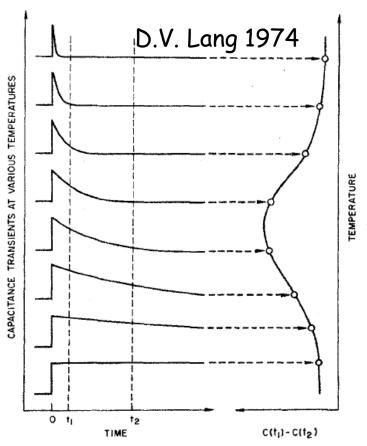


FIG. 6. Illustration of how a double boxcar is used to define the rate window. The left-hand side shows capacitance transients at varius temperatures, while the right-hand side shows the corresponding DLTS signal resulting from using the double boxcar to display the difference between the capacitance at time  $t_1$  and the capacitance at time  $t_2$  as a function of temperature.

# PICTS of ultra-pure germanium

JOURNAL OF APPLIED PHYSICS VOLUME 86, NUMBER 2 15 JULY 1999

# Photoinduced current transient spectroscopy of deep defects in n-type ultrapure germanium

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(Received 22 February 1999; accepted for publication 8 April 1999)

Photoinduced current transient spectroscopy (PICTS) is used to study deep minority carrier traps in n-type ultrapure germanium (shallow concentration of the order 10<sup>9</sup> cm<sup>-3</sup>). In this technique, which is a variant of deep level transient spectroscopy (DLTS), a neutral structure with two olunic contacts applied on two opposite faces of the sample is illuminated through one of the contacts with intrinsic, strongly absorbed light. The current transients which follow after interrupting the photoexcitation are analyzed using classical double lock-in DLTS resulting in the detection of centers in the minority half of the band gap (provided the back ohmic contact is negative). After correcting the PICTS spectra for the temperature dependence of the mobility, six peaks superimposed on a broad background are clearly resolved. The peaks are the same as the ones found earlier in high-purity n-type germanium (shallow concentration of the order 10<sup>10</sup> cm<sup>-3</sup>) using optical DLTS. These peaks are mainly Cu related. A formula to calculate concentrations from the PICTS spectra is deduced and verified experimentally. © 1999 American Institute of Physics. [80021-8979(99)02814-5]

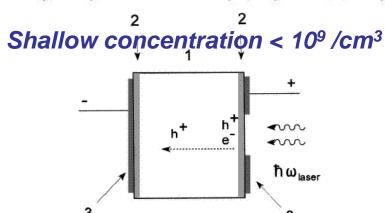


FIG. 2. Schematic configuration of the sample in PICTS of n-type ultrapure germanium. (1) Neutral semiconductor; (2) n<sup>+</sup> ohmic contact (Li diffused); (3) In foil.

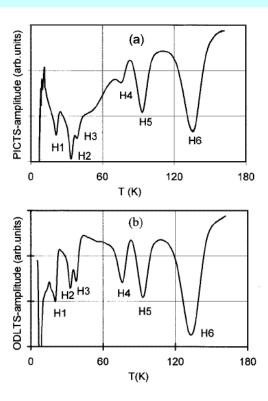


FIG. 5. (a) PICTS spectrum of *n*-type UP Ge [corresponding to Fig. 4(a)] corrected for the temperature dependence of the mobility,  $N_D$ – $N_A$   $\approx 10^9 \, \mathrm{cm}^{-3}$  [same measurement conditions as those in Fig. 4(a)]. (b) ODLTS spectrum of *n*-type HP Ge,  $N_D$ – $N_A$   $\approx 10^{10} \, \mathrm{cm}^{-3}$  [same measurement conditions as those in Fig. 4(a)].

# . . . ultra pure germanium

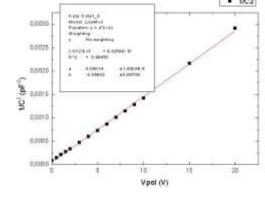
Blondeel (1999) ,Forment (2006)

What we are able to do now (DLTS limited to

specific samples)

TABLE II. Comparison of deep level concentrations calculated from ODLTS and PICTS measurements on four different kinds of HP and UP n-Ge samples. (-: No peak; ?: background difficult to subtract.)

	$N_D - N_A = 2 \times 10^9 \text{ cm}^{-3}$		$N_D - N_d = 2 \times 10^9 \text{ cm}^{-3}$	
	ODLTS (cm <sup>-3</sup> )	PICTS (cm <sup>-3</sup> )	ODLTS (cm <sup>-3</sup> )	PICTS (cm <sup>-3</sup> )
Cu <sub>3</sub> <sup>0/-</sup>	1.4×10 <sup>8</sup>	$1.4 \times 10^{8}$	4.8×10 <sup>7</sup>	$9.6 \times 10^{7}$
(Cu, H)6/-	$3.8 \times 10^{8}$	2.6×108	$1.1 \times 10^8$	$1.3 \times 10^8$
V2. H	1.8×10 <sup>5</sup>	$1.3 \times 10^{8}$	7.5×10 <sup>7</sup>	$5.1 \times 10^{7}$
Cu, H, Li	$1.1 \times 10^{6}$	$7.8 \times 10^7$	5.6×10 <sup>7</sup>	$2.8 \times 10^7$
(Cu, H)-/2-	2.5×10 <sup>8</sup>	$3.2 \times 10^{8}$	9.8×10 <sup>7</sup>	8.1×10 <sup>7</sup>
Cu. 22	$1.2 \times 10^{8}$	$5.0 \times 10^{8}$	$9.3 \times 10^{7}$	$8.2 \times 10^7$
	$N_D - N_A = 7 \times 10^9 \text{ cm}^{-2}$		$N_D\!-\!N_A\!=\!1\times10^{10}~{\rm cm}^{-3}$	
	ODLTS (cm <sup>-3</sup> )	PICTS (cm <sup>-3</sup> )	ODLTS (cm <sup>-3</sup> )	PICTS (cm <sup>-3</sup> )
Cu <sup>0</sup> -	-	-	7	1.0×10 <sup>8</sup>
(Cu, H) <sup>©-</sup>	-		2.1×10 <sup>8</sup>	2.5×108
V <sub>2</sub> , H	1.7×104	2.6×10 <sup>8</sup>	2.1×10 <sup>8</sup>	$2.8 \times 10^{8}$
Cu, H, Li	$3.4 \times 10^{8}$	2.1×10 <sup>8</sup>	7	$1.1 \times 10^{7}$
(Cu, H) - 2-	$1.0 \times 10^{1}$	5.9×107	1.3×10 <sup>8</sup>	$3.2 \times 10^8$
Cu <sub>2</sub> -'2-	5.8×10 <sup>8</sup>	4.9×10 <sup>8</sup>	$1.1 \times 10^{8}$	$3.7 \times 10^{5}$



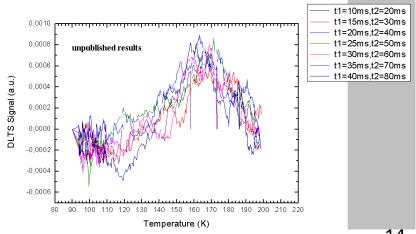
#### @77 K **HPGe** diode CP4 etched

DLTS spectrum with our setup on p-type (10<sup>11</sup> /cm<sup>3</sup>) germanium. Capacitance transients are digitized with a sampling scope (limited resolution < 11 bits) Noise and resolution are still unoptimized.

Implantation conditions of the studied metals

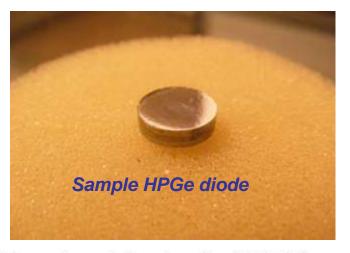
Metal	Implantation energy (keV)	Dose (at/cm <sup>2</sup> )	Treak (K)	$E_{\rm T}$ (eV)	$N_{\rm T}~({\rm cm}^{-3})$
Hf	190	$1 \times 10^{14}$	135	0.29	$9.0 \times 10^{11}$
Ti	90	5 × 1014	125	0.22	$1.3 \times 10^{14}$
Cr	90	$5 \times 10^{14}$	190	0.37	$1.3 \times 10^{14}$
Fe	90	$5 \times 10^{14}$	175	0.36	$2.4 \times 10^{14}$
Co	90	$5 \times 10^{14}$	155	0.33	$9.9 \times 10^{13}$
Ni	90	$5 \times 10^{14}$	95	0.16	$2.6 \times 10^{12}$
			160	0.30	$2.1 \times 10^{12}$

Annealing was performed at 500 °C for 5 min, except for the Ni-implanted sample that was annealed at 350 °C for 1 min. Also indicated is the peak position  $T_{peak}$ , the activation energy  $E_T$  and the trap concentration  $N_T$  derived from the DLTS analyses.



# Optical Deep Level Transient Spectroscopy (ODLTS) & Photoinduced Current Transient Spectroscopy (PICTS) of electronic defects in low-purity grade Ge





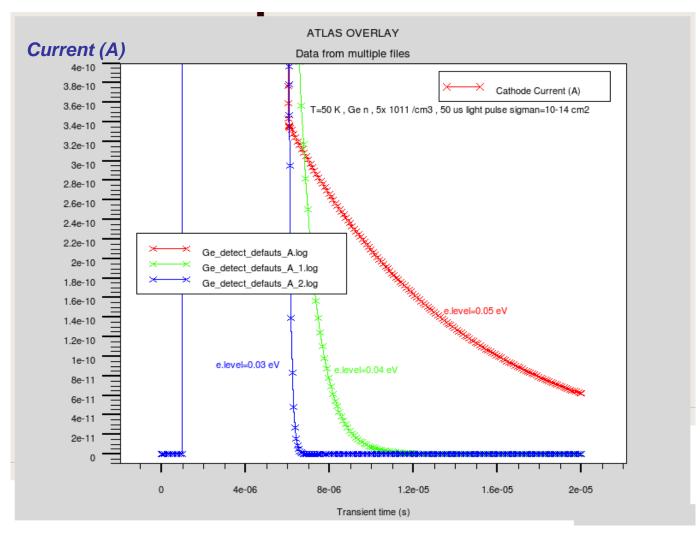
Experimental setup for PICTS/
DLTS under construction at CSNSM

DLTS operational but needs improvements

Oxford 23-09-2010

EURECA technical meeting, Munich Dec. 10, 2009

# PICTS (germanium, simulations)

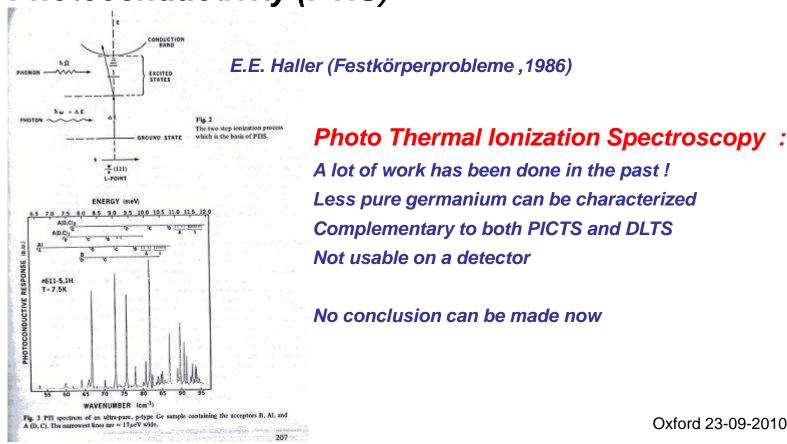


- Simulation based on a commercial software
- Current transients are slower when the level is deeper
- Dimensions :
- 300 µm thickness
- 1µm width
- 2 mm length
- Not a real detector

Oxford 23-09-2010

# Limitations ( with respect to PTIS)

- PICTS & DLTS, up to now: Investigations normally limited to deep traps
- Shallower levels can be detected using Far Infrared Photoconductivity (PTIS)



# What's proposed and under way

- PICTS on a real EDELWEISS detector
- Spectrum: "fingerprint" of the device
- Necessary to characterize detectors based on less pure Ge
- PICTS and DLTS setup under development at Orsay (first step LN2 next step LHe)
- Fast capacitance meter (CSNSM)
- Fast current amplifier (CSNSM)
- Temperature measurements (Pt resistor) (Saclay)
- LN2 (Saclay) and LHe (CSNSM) cryostats
- Pulse generator (CSNSM): need for a new pulse generator
- Pulsed LEDs (CSNSM)
- DAQ + Software (Saclay) need to buy an appropriate acquisition board + laptop computer