

Gibbs2 user's guide

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1 Introduction

Gibbs2 is a program for the calculation of the pressure and temperature dependence of the thermodynamic properties solid phases from ab initio data, within the framework of the quasiharmonic approximation. The predecessor of this code is gibbs, by M. A. Blanco, E. Francisco and V. Luaña, described in [\[orig\]](#).

In a typical calculation, for a single phase, the user selects a grid of volumes encompassing the equilibrium geometry. At those fixed volumes, the rest of the structural parameters are relaxed and a $E(V)$ curve is obtained, the static energy. In the simplest usage possible of gibbs2, this is the only necessary data with which to generate the thermodynamic properties at arbitrary pressures and temperatures. In the most complex, the full vibrational band structure at each volume is read, and the exact expressions of the quasiharmonic formulas are used. Therefore, one of the primary objectives of the program is fulfilled: giving the user a way to include as much information about the system as possible in order to improve the quality of the computed properties.

In a theoretical context, the effect of pressure is accounted for in a simple way by adding a $+pV$ term. The effect of temperature, however, requires a thermal model: a way of including the thermal contribution of the crystal degrees of freedom to the free energy. These contributions are dominated, in general, by the vibrational free energy, so the bulk of the gibbs2 code deals with how to incorporate the vibrational effects. Several thermal models with increasing complexity have been implemented:

- Static: no temperature effects.
- Debye and Debye-Gruneisen: require the knowledge of the static energy. Optionally, the Poisson ratio and the Gruneisen gamma can also be input.
- Debye-Einstein: in addition, requires the vibrational frequencies at the gamma point of the first Brillouin zone (1BZ).
- QHA: together with the static energy curve, either the phonon density of states or the frequencies on a grid sampling the 1BZ are required at each volume.

Once the thermal model is chosen, it is possible to build the vibrational Helmholtz free energy at any temperature (F_{vib}) and thus to find the equilibrium volume at a given T and p by minimizing the non-equilibrium Gibbs free energy:

$$G^*(V;p,T) = E(V) + pV + F_{\text{vib}}(V;T)$$

The equilibrium volume $V(p,T)$ is then used to compute the rest of the thermodynamic properties. Note that, in this formulation, the internal degrees of freedom (i.e. those that determine the geometry aside from the volume) are assumed to be unchanged by the vibrational effects.

Minimizing the G^* expression above requires finding the volume derivative of the Helmholtz free energy ($F(V;T) = E(V) + F_{\text{vib}}(V;T)$). Moreover, some thermodynamic properties require the computation of derivatives of the free energy or the entropy: for instance, the bulk moduli, Gruneisen parameter,... Therefore, an analytic expression (a thermal equation of state, EOS) that fits both $E(V)$ and $F(V;T)$ is required, which needs to be good enough to reliably yield interpolated values of the energy and its derivatives.

A great deal of work has been put into the robust determination of energy-volume fits from which sensible interpolations and derivatives can be calculated. The recommended procedure involves performing linear fits of polynomials in a chosen strain (Birch-Murnaghan, Poirier-Tarantola,...) with increasing degree, and then obtaining an average polynomial. The averaging procedure provides a statistical measure of the goodness of the fit in the form of calculated error bars of the properties. We have found this method extremely robust, capable of smoothing out noise, detecting outliers and giving a good estimation of the general quality of the input data. In addition, gibbs2 is also able to fit the $E(V)$ to other traditional EOS like Vinet or Holzapfel's AP2 using a non-linear minimization algorithm. Note that polynomial averages are excellent for interpolation on a grid of volumes, but totally fail to reproduce a solid behavior outside of the fitted volume grid. In such a case, the volume grid needs to be extended or a conventional EOS can be used. In the sense of fits, gibbs2 is more or less a fortran90 reimplementation of the asturfit companion package ([\[fit1\]](#) and [\[fit2\]](#)).

Using the experimental volume (and optionally bulk modulus), the static energy can be corrected to remove the known systematic trends present in DFT calculations. This is called the empirical energy correction (EEC). Several EEC have been implemented, and will be described in a future work.

In addition to the vibrational contribution, other free energy terms can be included using gibbs2. Namely, the electronic contribution in metals, by reading a polynomial fit to the temperature-dependent free energy calculated using Mermin's approach, or by using a free electron model. Other contributions (magnetic, configurational) are easy to implement in the current scheme of the code.

Sometimes it is interesting to compare the Gibbs free energies of different phases of the same solid to determine its phase diagram. Gibbs2 accepts an arbitrary number of phases for the same compound and calculates the most stable at any

pressure and temperature. The temperature-dependent transition pressures are calculated. In fact, the input of phases (through the PHASE keyword) is even more general: several different thermal and electronic models, fitting expressions, energy corrections, etc. can be used for each phase in the same run.

2 2 Installation

The gibbs2 distribution contains the following files and directories:

- src/ - Source code of the program (see below).
- doc/ - User's guide in plain text (gibbs2.txt) and PDF (gibbs2.pdf). The user's guide can be compiled using compile.sh, but the compiled pdf is given together with the package. The compilation requires a working LaTeX installation and the reStructuredText package.
- tests/ - A collection of test cases that can be also used as templates. We have tried to cover all possible normal uses of the code.
- dat/ - Data for the tests. Volume, density, phonon density of states and electronic free energy have been calculated for MgO (phases B1 and B2), diamond and fcc Al; LDA and PBE xc functionals.
- ChangeLog - A summary of recent changes.
- README - A short orientational note.

To install the program, enter the src/ directory and edit the Makefile.inc file. This file is included in the actual Makefile and contains the description of the compiler and its options. By default, the GNU fortran compiler (gfortran) is used. The uncommented section should look like this:

```
1 # The GNU fortran compiler (gfortran)
2 ifeq ($(DEBUG),1)
3   FC = gfortran
4   FCFLAGS = -O -g -fbounds-check -Wall -Wunused-parameter
5   -ffpe-trap=invalid -fbacktrace -fdump-core
6   LDFLAGS =
7 else
8   FC = gfortran
9   FCFLAGS = -O
10  LDFLAGS =
11 endif
```

The variable FC is the name of the fortran90 compiler, FCFLAGS are the compiler flags and LDFLAGS the linker flags. Two sets of options can be used: for a debug compilation (first block) and a normal compilation (second block). Once the Makefile.inc is ready, compile gibbs2 using:

```
1 make
```

The compilation generates a gibbs2 binary that can be moved or linked into your \$PATH. To compile with the DEBUG option activated, use:

```
1 make debug
```

The name of the debug binary file is gibbs2_dbg. The compiled debug code is exactly the same (no preprocessor directives).

Other make options are:

```
1 make clean
2 make veryclean
3 make mrproper
```

These remove the objects (first), objects and static libraries (second) and objects, libraries and binaries (third).

3 3 Usage

Gibbs2 reads one main input file (the `ing` file, although the extension is not mandatory) and generates the main output that is written to `stdout` and several auxiliary files, as will be described below. The output files have the same root (that is, the name up to the last `.`) as the input. The program is run as:

```
1 gibbs2 [options] [file.ing [file.out]]
```

If `file.ing` is present, the input is read from that file. In any other case, the standard input is used. The same applies for `file.out`: if it is present, output is redirected to it. Otherwise, the `gibbs2` output is written to the standard output. As a corollary, `gibbs2` can also be used:

```
1 gibbs2 [options] < file.ing > file.out
```

The command line options are described now. Each of them corresponds to a SET keyword in the input file, which is indicated in parentheses.

- **-n,--noplots** Inhibits the creation of all the auxiliary files written by `gibbs2`. The only output written goes to `stdout` and `stderr`. (SET WRITELEVEL 0)
- **-e,--eos** Same as `-n`, but the `eos` (dynamic equation of state) and `eos_static` (static equation of state) files are written. (SET WRITELEVEL 1)
- **-b,--errorbar** Calculate and output the error bars for each thermodynamic quantity. The error values are marked by an "e" at the beginning of the line in the `eos` file. (SET ERRORBAR)
- **-t,--notrans** Do not compute transition pressures. (SET NOTRANS)
- **-d,--noplotdh** Do not produce plots of enthalpy differences to the first phase in input. (SET NOPLOTDH)
- **-f,--noefit** Do not produce plots of the input and fitted static energy. (SET NOEFIT)
- **-h,--help,-?** Command-line help.

4 4 The structure of the input and output files

4.1 4.1 A simple example

Gibbs2 reads its input from a single file, the `ing` file. This file contains keywords that are parsed by the program: the full list is described below. Comments can be included using the `#` symbol as the first non-blank character of the line. Lines can be continued using a backslash (`\`) as the last character.

Let us start with a simple example to introduce the basic keywords. For instance, the following is almost the simplest `ing` file possible, containing data from a LDA PP+PW calculation of `MgO`:

```
1 mm 40.3044
2 vfree 2
3 phase mgo
4 81.8883583665837 -73.5171659350000
5 86.0358791612784 -73.5360133400000
6 90.1833999559730 -73.5508544000000
7 94.3309207506677 -73.5624133500000
8 98.4784415453624 -73.5712754950000
9 102.6259623400570 -73.5779108750000
10 106.7734831347517 -73.5826963000000
11 110.9210039294464 -73.5859397750000
12 115.0685247241411 -73.5878968050000
13 119.2160455188357 -73.5887770750000
14 123.3635663135304 -73.5887555350000
15 127.5110871082251 -73.5879775400000
16 131.6586079029198 -73.5865593900000
17 135.8061286976144 -73.5846083150000
18 139.9536494923091 -73.5822069800000
19 144.1011702870038 -73.5794278200000
20 148.2486910816984 -73.5763274850000
```

```

21 152.3962118763931    -73.5729658900000
22 156.5437326710878    -73.5693791200000
23 160.0000000000000    -73.5662512150000
24 endphase

```

The tokens mm, vfree, phase and endphase are keywords: special words that are parsed by the program. MM is the molecular mass of the primitive cell in amu. In the case MgO, Mg is 24.3050 amu and O is 15.9994, and one of each atoms form the primitive fcc unit cell. VFREE is the total number of atoms in the primitive cell. The remaining keyword, PHASE, is the most important in gibbs2: each time it appears, a new phase of the considered solid is added for computation. Thus, if we were interested in the B1 -> B2 transition of MgO, we would have something like:

```

1 [...]
2 phase mgo:B1
3 ...
4 endphase
5 phase mgo:B2
6 ...
7 endphase

```

and gibbs2 would actually compute transition pressures, the phase diagram, the thermodynamic properties of the stable phase,... If, for instance, we are interested in comparing different temperature models, we could do:

```

1 [...]
2 phase mgo:debye tmodel debye
3 ...
4 endphase
5 phase mgo:debeins tmodel debye_einstein
6 ...
7 endphase

```

to compare the thermodynamic properties calculated by the Debye model or the Debye model with added Einstein frequencies. The same principle applies if we want to compare different equations of state, energy corrections,... Thus, the PHASE keyword represents a generalized phase of the solid given by our model representation, not just a physical phase. The string following the PHASE keyword (mgo, mgo:debye, etc.) is the identifier of that phase in gibbs2: it is used to label the phase in output and plots. Each phase is also labeled with an integer, that is the order in which it appears in the input. In the input above, mgo:debye is phase 1 and mgo:debeins is phase 2.

Let us go back to the simple example. As hinted above, the phase keyword accepts multitude of options, but we have opted for simply assuming the sensible defaults: the Debye model for temperature effects, Birch-Murnaghan polynomial averages,... A detailed account of these defaults is given in the reference section. The PHASE keyword opens an environment in the input that ends with the ENDPHASE keyword. Each record of the environment corresponds to one data point: the first field is the volume and the second is the energy. The default input and output units in gibbs2 are atomic units, so volumes are in bohr³ and energies are in Hartree.

This input is saved to a file (say, mgo.ing) and run using:

```

1 gibbs2 mgo.ing mgo.out

```

The run time is brief (less than 1 second using my 2GHz cpu). Gibbs2 generates some output files, all using the same root as the input file (that is, mgo):

```

1 mgo.out
2 mgo_all_p.gnu
3 mgo_all_t.gnu
4 mgo.efit
5 mgo_efit.aux
6 mgo_efit.gnu
7 mgo.eos
8 mgo.eos_static

```

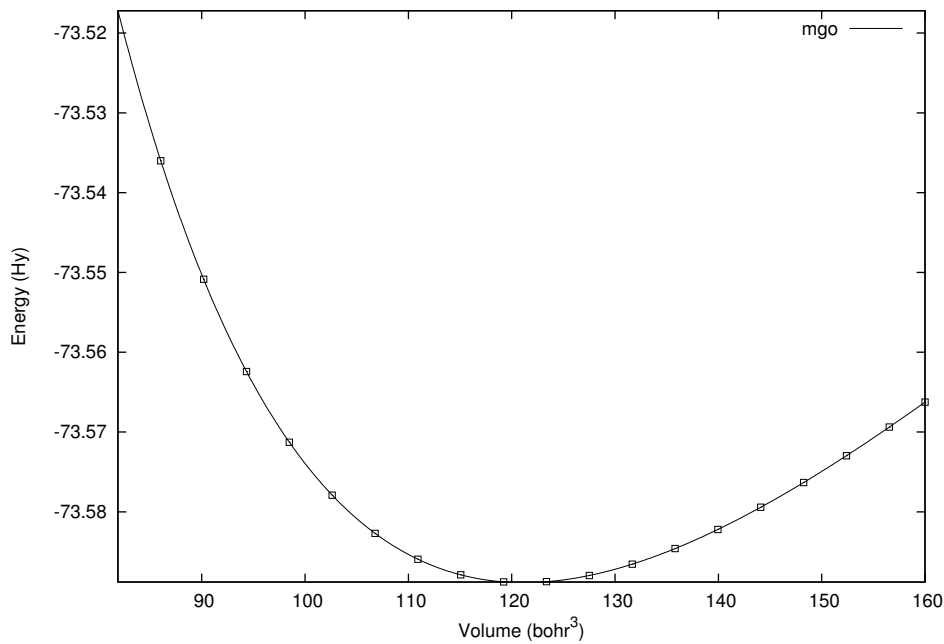
The mgo.out file contains the main output, and we comment it below. The files mgo.efit, mgo_efit.aux and mgo_efit.gnu are plots of the fitted static energy vs. volume together with the original data. To obtain the plot, simply do:

```

1 gnuplot mgo_efit.gnu

```

I am using gnuplot version 4.4 (probably any later or previous version down to 4.2 will do). The script requires the epstopdf and pdfcrop programs, easily obtainable from the net or your distribution's software repository. Gnuplot generates a pdf file containing the plot:



The points represent the original data and the solid line is a 12-th degree polynomial in the Birch-Murnaghan strain. The fit is apparently excellent.

Another important file is mgo.eos_static, containing the static properties derived from the E(V) curve (the first lines have been wrapped, the rest have been trimmed for clarity):

```
1 # Lines beginning with 'e' contain fit error estimation.
2 # Phase mgo
3 #      p (GPa)      E (Ha)      V(bohr^3)  V/V0      p_fit (GPa) \
4 #              B (GPa)      Bp      Bpp (GPa-1)
5      0.0000 -7.358887030E+01 121.1510 1.00000000 0.0000 \
6              171.8631 4.1045942 -2.3291E-02
7 e 0.0000 0.000000000E+00 0.0016 0.0000130 0.0000 \
8              0.0325 0.0089770 8.0853E-04
9      5.0505 -7.356836447E+01 117.8343 0.9726230 5.0505 ...
10 e 5.0505 0.000000000E+00 0.0016 0.0000134 0.0000 ...
11      10.1010 -7.354839133E+01 114.9271 0.9486271 10.1010 ...
12 e 10.1010 0.000000000E+00 0.0014 0.0000118 0.0000 ...
13 [...]
14      141.4141 -7.312371446E+01 82.9321 0.6845349 141.4141 ...
15 e 141.4141 3.932099939E-06 0.0023 0.0000191 0.0000 ...
16      146.4646 -7.310953178E+01 82.3100 0.6793997 146.4646 ...
17 e 146.4646 2.697398305E-06 0.0144 0.0001189 0.0000 ...
```

The calculated properties are the static energy, volume, compression, bulk modulus and its pressure derivatives for a set of pressures. This list of pressures can be controlled using the keyword *pressure* (see below). By default, gibbs2 calculated 100 pressure data points from 0 GPa to the maximum pressure allowed by the static energies, or 500 GPa. The lines beginning with an 'e' contain the error of the fit, as estimated by the polynomial averaging technique.

The file mgo.eos contains the relation of the thermodynamic properties calculated for the pressure and temperature conditions given in input. For instance, in our example,

```
1 # Phase mgo
2 # 01:p(GPa) 02:T(K) 03:V(bohr^3) 04:Estatic (Ha) 05:G (kJ/mol)
3 # 06:Gerr(kJ/mol) 07:p_sta(GPa) 08:p_th(GPa) 09:B (GPa) 10:U-Esta (kJ/mol)
4 # 11:Cv(J/molK) 12:F-Esta (kJ/mol) 13:S(J/molK) 14:ThetaD(K) 15:gamma
5 # 16:alpha(10^-5/K) 17:dp/dT(GPa/K) 18:Bs (GPa) 19:Cp(J/molK) 20:B_Tp
6 # 21:B_Tpp(GPa-1) 22:Fvib(kJ/mol) 23:Fel(kJ/mol) 24:Uvib(kJ/mol) 25:Uel(kJ/mol)
7 # 26:Svib(J/molK) 27:Sel(J/molK) 28:Cv_vib(J/molK) 29:Cv_el(J/molK)
8 #      p (GPa)      T (K)      V(bohr^3)      Estatic (Ha)      G (kJ/mol)      ...
9      0.0000      0.00      123.0980      -7.358878E+01      -1.9319213537E+05 ...
10      5.0505      0.00      119.6117      -7.358881E+01      -1.9313745955E+05 ...
```



```

11      10.1010      0.00      116.5693      -7.358833E+01      -1.9308425053E+05 ...
12      ...
13      141.4141      0.00      83.6628      -7.352577E+01      -1.9195765650E+05 ...
14      146.4646      0.00      83.0679      -7.352298E+01      -1.9192008494E+05 ...
15
16
17      0.0000      6.57      123.0980      -7.358878E+01      -1.9319213537E+05 ...
18      ...
19      146.4646      6.57      83.0679      -7.352298E+01      -1.9192008494E+05 ...
20
21      ...
22
23      0.0000      650.12      125.8789      -7.358837E+01      -1.9321010456E+05 ...
24      ...
25      146.4646      650.12      83.3617      -7.352437E+01      -1.9192739359E+05 ...

```

The file is organized in successive blocks, one for each phase in the input starting with the line '# Phase' followed by the identifier of the phase. After it, there is a header that specifies the thermodynamic properties calculated and the field they occupy. The data is organized in smaller blocks, one for each temperature, that are separated by two blank lines (useful when using 'index' in gnuplot, for example). These smaller blocks are composed of long records each of them corresponding to the pressure and temperature indicated in the first and second fields. The rest of the fields are the calculated properties, as indicated in the header. As with the pressure, the temperatures for which the properties are calculated can be specified in the input, but in this simple example the default list is used: 100 temperature data points from 0 to 1.5 times the maximum Debye temperature.

The calculated properties are:

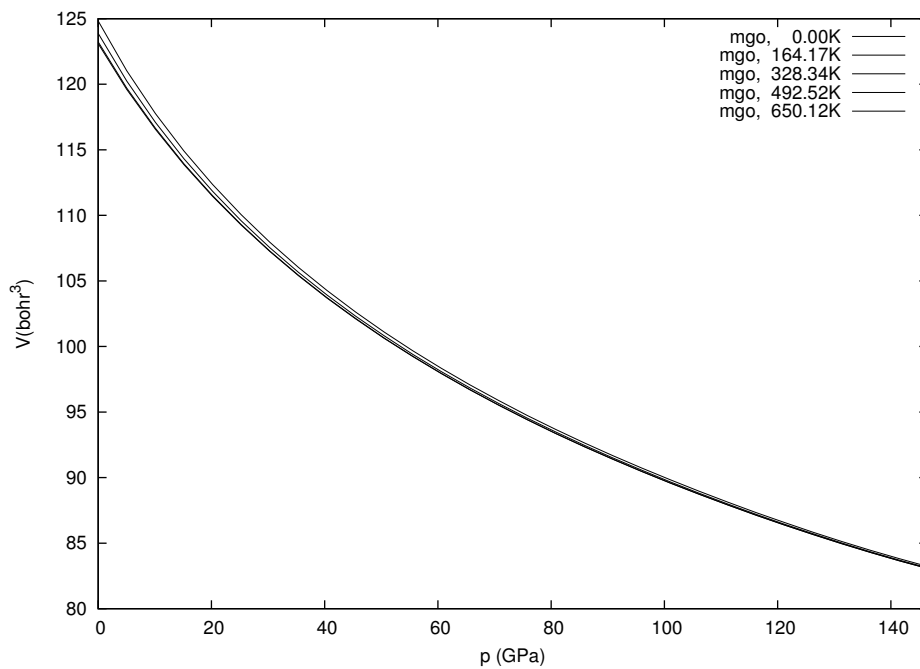
- **01:p(GPa)** Pressure (given by the user).
- **02:T(K)** Temperature (given by the user).
- **03:V(bohr^3)** Equilibrium volume, $V(p,T)$.
- **04:Estatic(Ha)** Static energy, interpolated at the equilibrium volume.
- **05:G(kJ/mol)** Gibbs free energy, $G = E_{\text{static}} + pV + F_{\text{vib}} + F_{\text{el}} + \dots$.
- **06:Gerr(kJ/mol)** The Helmholtz free energy contained can be calculated in two ways: computing it directly using the equilibrium volume or by interpolation from the fit. Gerr is the difference between both approaches, and is usually negligible.
- **07:p_sta(GPa)** Static pressure, $-dE_{\text{static}}/dV$.
- **08:p_th(GPa)** Thermal pressure, $-(d(F-E_{\text{static}})/dV)_T$. Note that $p = p_{\text{sta}} + p_{\text{th}}$.
- **09:B(GPa)** Isothermal bulk modulus, $B_T = -V \cdot (dp/dV)_T = V \cdot (d^2F/dV^2)_T$.
- **10:U-Esta(kJ/mol)** Non-static contribution to the internal energy: $U_{\text{vib}} + U_{\text{el}} + \dots$.
- **11:Cv(J/molK)** Constant volume heat capacity, $C_v = (dU/dT)_V = C_{v,\text{vib}} + C_{v,\text{el}} + \dots$.
- **12:F-Esta(kJ/mol)** Non-static contribution to the Helmholtz free energy: $F_{\text{vib}} + F_{\text{el}} + \dots$.
- **13:S(J/molK)** Entropy: $S_{\text{vib}} + S_{\text{el}} + \dots$.
- **14:ThetaD(K)** Debye temperature, only calculated for Debye-like models.
- **15:gamma** Gruneisen parameter, $\gamma = \alpha \cdot B_T \cdot V / C_v = V / C_v \cdot (dS/dV)_T$.
- **16:alpha(10^-5/K)** Volumetric thermal expansion coefficient, $\alpha = 1/V \cdot (dV/dT)_p = \gamma \cdot C_v / V / B_T$.
- **17:dp/dT(GPa/K)** $(dp/dT)_V = \alpha \cdot B_T$.
- **18:Bs(GPa)** Adiabatic bulk modulus $B_S = -V \cdot (dp/dV)_S = V \cdot (d^2U/dV^2)_S = B_T \cdot (1 + \alpha \cdot \gamma \cdot T)$.
- **19:Cp(J/molK)** Constant pressure heat capacity, $C_p = (dH/dT)_p = C_v \cdot (1 + \alpha \cdot \gamma \cdot T)$.
- **20:B_Tp** Isothermal pressure derivative of the isothermal bulk modulus, $(dB_T/dp)_T$.
- **21:B_Tpp(GPa-1)** Isothermal second pressure derivative of the isothermal bulk modulus, $(d^2B_T/dp^2)_T$.

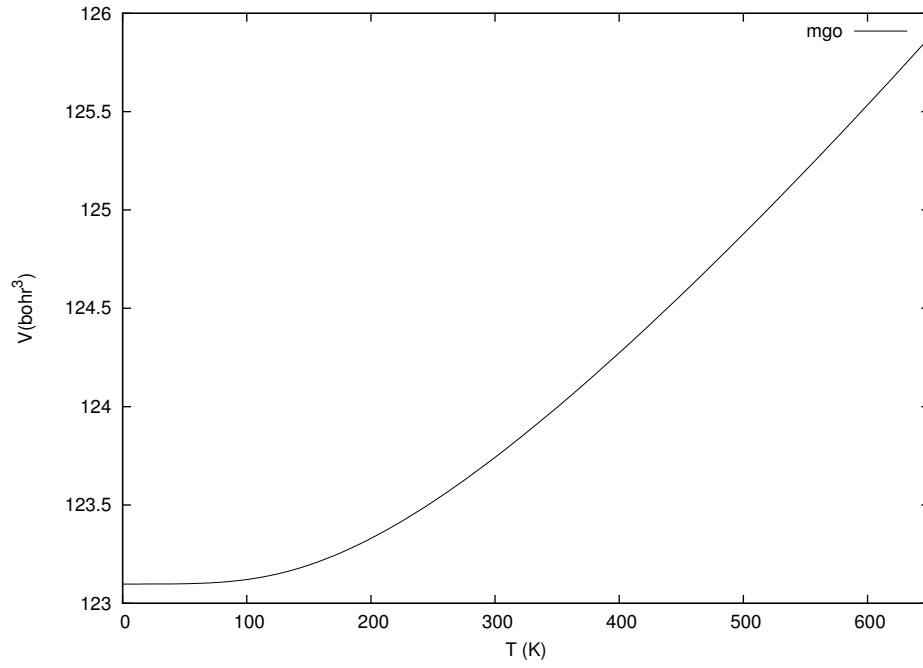
- **22:Fvib(kJ/mol)** Vibrational contribution to the Helmholtz free energy.
- **23:Fel(kJ/mol)** Electronic contribution to the Helmholtz free energy (if no electronic contribution model is used, 0).
- **24:Uvib(kJ/mol)** Vibrational contribution to the internal energy.
- **25:Uel(kJ/mol)** Electronic contribution to the internal energy.
- **26:Svib(J/molK)** Vibrational contribution to the entropy.
- **27:Sel(J/molK)** Electronic contribution to the entropy.
- **28:Cv_vib(J/molK)** Vibrational contribution to the constant volume heat capacity.
- **29:Cv_el(J/molK)** Electronic contribution to the constant volume heat capacity.

The two gnuplot scripts `mgo_all_p.gnu` and `mgo_all_t.gnu` are simple ways of visualizing the information contained in the `mgo.eos` file. To run them:

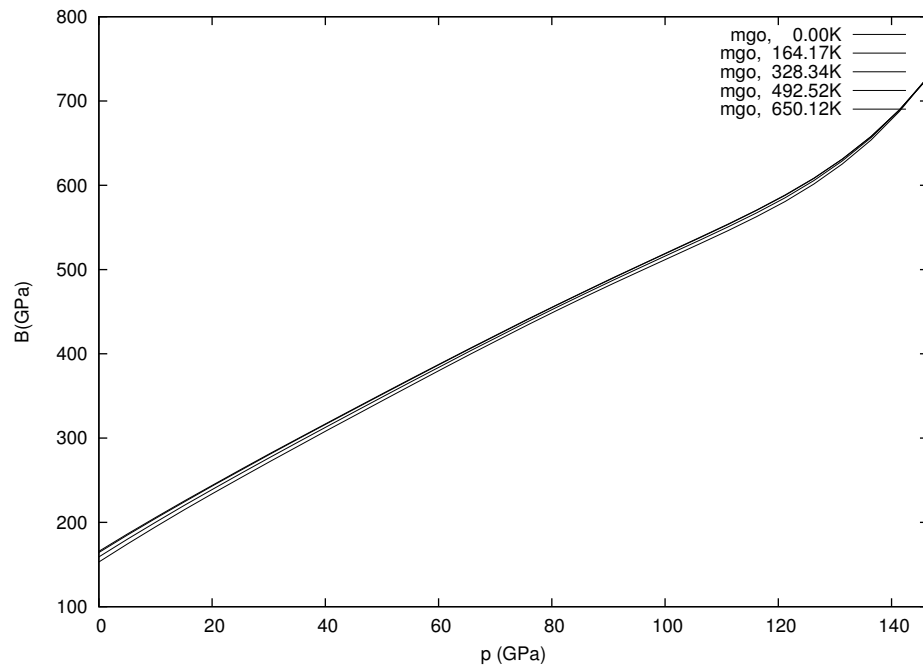
```
1 gnuplot mgo_all_p.gnu
2 gnuplot mgo_all_t.gnu
```

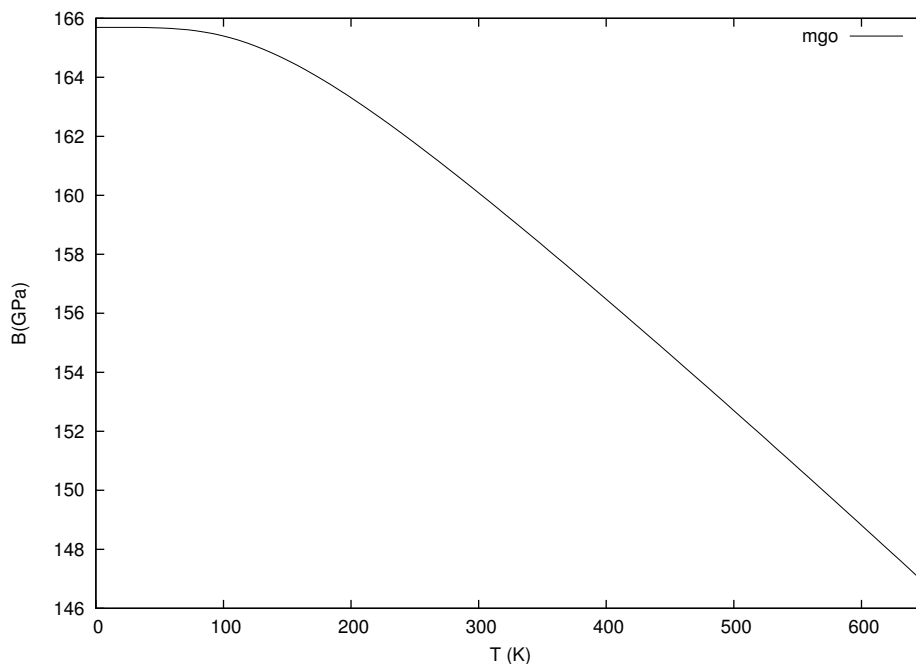
This creates a good number of pdf files called `mgo_p_xx.pdf` and `mgo_t_xx.pdf`. The `mgo_p_xx.pdf` file represents the isotherms of the `xx` property versus pressure, where `xx` is one of the numeric labels above. The `mgo_t_xx.pdf` represent the temperature dependence of the `xx` property at $p = 0$ GPa. For our example, `mgo_p_03.pdf` are the calculated $V(p)$ isotherms and `mgo_t_03.pdf` is a plot representing thermal expansion at zero pressure:





Similarly, mgo_p_09.pdf contains several curves representing the pressure dependence of the bulk modulus at the computed temperatures and mgo_t_09.pdf is a plot of $B_T(T)$ at 0 pressure:





The standard output file generated by gibbs2, mgo.out, contains useful information and a detailed relation of the auxiliary output files. It is structured in blocks denoted by a header that starts with an asterisk. For instance, the first block (Input) contains information about the input values of some general variables, and it is self-explanatory:

```
1 * Input
2   Title:
3   Output file (lu= 2): stdout
4   Units: output is in atomic units, except where noted.
5   Number of atoms per formula: 2
6   Molecular mass (amu): 40.30440000
7   ...
```

Following this section, the pressure range is displayed. We have used the default pressures, so gibbs2 automatically sets this list to 100 points from 0 to the maximum pressure attainable with the input data, p_max. This p_max is calculated as the slope at the point of the volume grid with smallest volume: any pressure higher than p_max would preclude the appearance of the minimum of the static enthalpy on the volume grid:

```
1 * Pressure range examined
2   Min_phases{p_max} (GPa): 118.529
3   Pressure range (GPa): 0.000 -> 500.000
4   Number of p points: 100
5   New pressure range (GPa): 0.000 -> 146.465
6   New number of p points: 30
```

Next, the information about the phases is presented, including static equilibrium properties, statistical measures about the quality of the static fit, and the static equation of state, much like in the mgo.eos_static file:

```
1 * Phase information after initial setup
2 + Phase 1 (mgo)
3   Number of volume points: 20
4   p(V) input data? F
5   Pressure range (GPa): -27.115 -> 118.529
6   Number of interpolated fields : 0
7   Input units:
8     Volume : bohr^3
9   ...
10  Temperature model: Debye, Td from static B(V).
11  All data points are ACTIVE for dynamic calculations
12
13  Fit to static E(V) data:
14  # Copy in file : mgo.eos_static
15  # Lines beginning with 'e' contain fit error estimation.
16  #   p (GPa)      E (Ha)      V(bohr^3)      V/V0      p_fit (GPa)      B (GPa)
17  Bp              Bpp (GPa-1)
```

```

17      0.0000 -7.358887030E+01 121.1510 1.0000000 0.0000 171.8631 4.1045942
    -2.3291E-02
18 e      0.0000 0.000000000E+00 0.0016 0.0000130 0.0000 0.0325 0.0089770
    8.0853E-04
19 ...

```

The static energy plots are calculated next, and the Debye temperatures are computed. The default thermal model is the Debye model in the Slater formulation: the Debye temperatures, $T_d(V)$ are computed from the static bulk moduli $B(V)$, and the Poisson ratio is assumed to be volume independent, and equal to 1/4. This information is presented in the output:

```

1 * Computed Debye temperatures from static data
2 + Phase mgo
3 # ThetaD at static eq. volume: 836.45
4 # V(bohr^3) Tdebye(K) Tdebye_slater(K)
5 81.8884 1575.07 1575.07
6 86.0359 1460.24 1460.24
7 90.1834 1364.07 1364.07
8 94.3309 1274.15 1274.15
9 ...
10 160.0000 433.41 433.41

```

The minimum of $T_d(V)$ is usually found at the largest volume. This T_{d_min} is used to set the default temperature datasets used in the dynamic calculation:

```

1 * Temperature range examined
2 Min_{DebyeT} (K): 433.415
3 Temperature range (K): 0.000 -> 650.122
4 Number of T points: 100

```

Once the static calculation is over, gibbs2 loops over temperatures and pressures and calculates the thermodynamic properties for each pT point, and writes the results to the eos file:

```

1 * Calculated temperature effects
2 Writing file : mgo.eos

```

A final note: it is usually convenient to keep the gibbs2 input file separated from the actual phase data. The keywords *file* and *using* (or *u* for short) can be used in *phase* to read an external file containing the volume and energy data. For instance, mgo.ing could be:

```

1 mm 40.3044
2 vfree 2
3 phase mgo file mgo.dat u 1:3

```

and the E(V) data would be read from the fields 1 and 3.

4.2 Fitting equations of state

We continue with a slightly more complex example. The following input file reads an external file (c.dat) containing the energy-volume curve of diamond (PBE), and applies several different fits to the data, using the Debye-Gruneisen temperature model. Each equation of state examined corresponds to one **phase** keyword: the dataset is always the same, but the model representation given by the equation of state used to fit the data changes.

The input file is:

```

1 set notrans
2 mm 24.0214
3 vfree 2
4 pressure 0
5 temperature 0 10 3000
6
7 phase BM3 file c.dat tmodel debye_gruneisen dm \
8     units energy ry \
9     fit strain bm 3
10 phase BM4 file c.dat tmodel debye_gruneisen dm \
11     units energy ry \

```

```

12         fit strain bm 4
13 phase PT3 file c.dat tmodel debye_gruneisen dm \
14         units energy ry \
15         fit strain pt 3
16 phase PT4 file c.dat tmodel debye_gruneisen dm \
17         units energy ry \
18         fit strain pt 4
19 phase Vinet file c.dat tmodel debye_gruneisen dm \
20         units energy ry \
21         fit vinet
22 end

```

and the data file (c.dat) is:

```

1 40.000000000000000 -22.107087940000000
2 41.666666666666664 -22.204061660000001
3 ...

```

Note the first lines in the input are preceded by '#': this means these lines are comments, and therefore are ignored by gibbs2. Note also the backslashes that continue **phase** lines. The **notrans** option is activated with a **set** keyword. **set** keywords modify the global behavior of gibbs2: an exhaustive relation of these options can be found in the reference section of this manual. The **notrans** option deactivates the calculation of the phase diagram (transition pressures, most stable phase,...). It is clear that in this case it is not meaningful. The **mm** and **vfree** keywords are the same as in the previous example: the primitive cell of diamond contains 2 atoms, and the molecular weight is 2 times the mass of a carbon atom. The thermodynamic properties are calculated only at null pressure, and in a temperature grid spanning from 0 to 3000 K in steps of 10 K.

Five phases are defined, each of them using the same dataset (contained in the file c.dat), which is interpreted to be a list of volumes (in bohr³, the default) and energies in Ry, because of the **units** keyword. All of them use the same temperature model (Debye-Gruneisen with Dugdale-McDonald gamma) and different equations of state. The equations of state used are Birch-Murnaghan (3rd and 4th order), Poirier-Tarantola (3rd and 4th order) and Vinet. All of them carry out linear fits of the energy versus some strain except in the case of Vinet. This and other technical details of the fit are treated below, in the reference for the **fit** keyword.

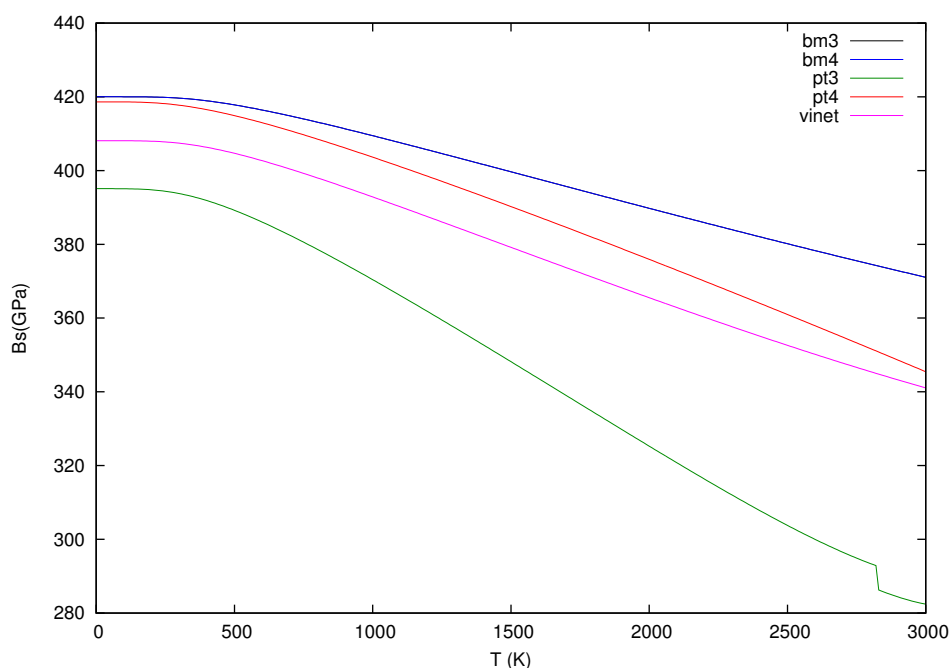
The structure of the output is more or less the same as in the previous example, but each phase receives its own block inside each of the sections. Plotting the temperature dependence of the properties with:

```

1 gnuplot c_all_t.gnu

```

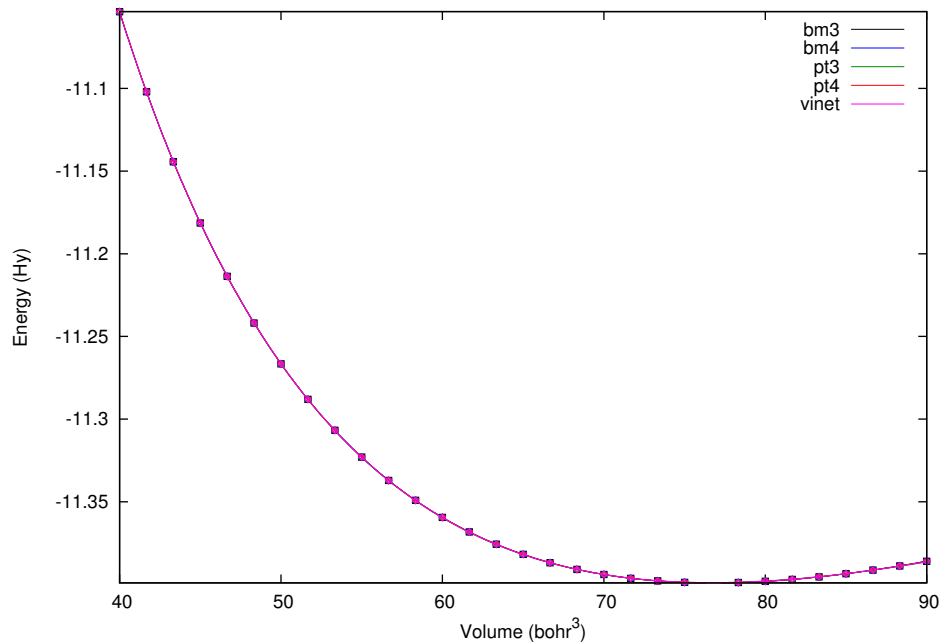
yields, for instance, a B_s(T) plot like the following:



Judging from this picture, it is clear that choosing a good equation of state is very important. At 2000 K, the range spanned by different equations of state is as large as 80 GPa. The recommended procedure in gibbs2 is using an average of polynomials in a given strain (for instance, a BM strain). This method, in addition, provides a good estimation of the quality of the fit

by means of error bars of the calculated properties. It is interesting to note that, in spite of the dramatic differences between EOS in the $B_S(T)$ curve, the energy-volume curve is superbly represented by *all* the EOS used:

```
1 gnuplot c_efit.pdf
```



Unfortunately, this plot is too often a criterion of the quality of a fit in the literature.

It is important to note that `gibbs2` is not designed for extrapolation. This means that the thermodynamic properties are never calculated on equilibrium volumes outside of the input volume grid.

A final note: in cases when the input dataset contains few points, the default maximum strain polynomial degree (`mpar = 12`) may be inadequate and cause an overfitting problem. In such cases, it is wise to use the `set mpar` keyword (see below) to set this parameter to a lower value.

4.3 4.3 Temperature models

Gibbs2 implements several models for the inclusion of temperature effects to the results of an ab initio calculation. The models provided cover a range of complexity (and accuracy), with the Debye model in Slater's implementation being the most simple and the full quasiharmonic approximation being the most complex. Likewise, more complex temperature models require that the user provides larger quantities of data: for instance, the Debye model requires only the input of the static $E(V)$ curve (and optionally the experimental Poisson ratio), while the QHA model requires the phonon spectrum calculated at each grid volume. The Debye-Grüneisen model is an improvement upon the simple Debye model. The quasiharmonicity introduced by assuming that the Poisson ratio does not change with volume is corrected by assuming an approximate Grüneisen gamma, such as Dugdale-McDonald's formula. With it, it is possible to get a more realistic evolution of the Debye temperature parameter than in the original Slater's model.

Also, in molecular crystals, it is usually necessary to account for the (probably many) optic branches of the spectrum. To that end, the Debye-Einstein model reads the optic frequencies at gamma and builds a phonon spectrum using a Debye model for the acoustic part, and a sum of Dirac deltas for the optic part, centered around the Gamma frequencies. The temperature models are explained in full in the reference section and in the companion articles.

Let us illustrate the use of the various models with a simple example. The following input compares different models on the B1 phase of MgO (calculated with PP+PW, PBE xc functional).

```
1 set notrans
2 mm 40.3044
3 vfree 2
4 pressure 0 1 250
5 temperature -1
6
7 phase debye file ../dat/mgo_pbe/mgo.res tmodel debye \
8             units energy ry freq cm-1 \
9             prefix ../dat/mgo_pbe/
10 phase debgrun file ../dat/mgo_pbe/mgo.res tmodel debye_gruneisen dm \
```

```

11         units energy ry freq cm-1 \
12         prefix ../dat/mgo_pbe/
13 phase debeins file ../dat/mgo_pbe/mgo.res tmodel debye_einstein \
14         units energy ry freq cm-1 \
15         prefix ../dat/mgo_pbe/
16 freqg0 debeins
17     402.9580 402.9580 701.1656
18 endfreqg0
19 phase qha file ../dat/mgo_pbe/mgo.res tmodel qha \
20         units energy ry freq cm-1 \
21         prefix ../dat/mgo_pbe/
22 end

```

In this case, we calculate the pressure dependence of the thermodynamic properties with pressure (up to 250 GPa) at room temperature (**temperature** -1). Each model corresponds to a **phase** keyword, all of them using the same dataset and even the same input file, consisting of:

```

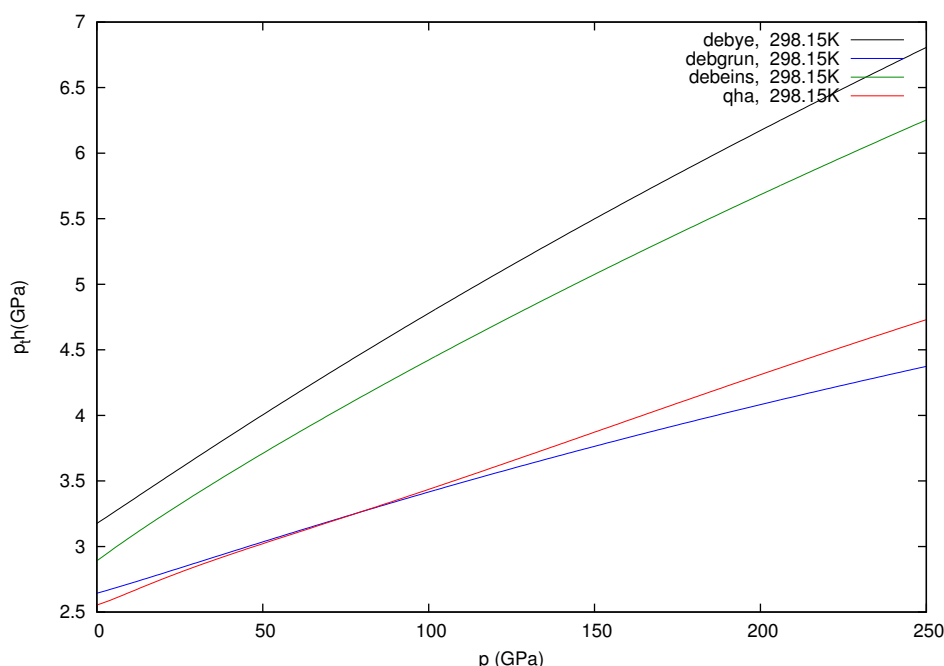
1 40.000000000000000 -145.125645270000007 001_/001_.phdos
2 40.700434586218826 -145.213514589999988 002_/002_.phdos
3 ...
4 160.000000000000000 -147.254963110000006 129/129.phdos

```

Depending on the model, some or all of these fields are read. In the Debye model, the fields 1 and 2 (volume and energy) are read and the third field that points to the phonon DOS is ignored. Note that the files containing the phonon DOS are located in a subdirectory 'xxx/' relative to the directory of the input data file, which in turn is located in '../dat/mgo_pbe/' relative to the gibbs2 input file. To make gibbs2 find the phonon DOS files, it is necessary to include a 'prefix' keyword, whose argument is prepended to the phonon DOS file names. For instance, for the first file, '001_/001_.phdos', prepending the prefix '../dat/mgo_pbe/' yields the file '../dat/mgo_pbe/001_/001_.phdos', which is the correct path relative to the gibbs2 input file. It is also possible to use absolute path prefixes.

The Debye-Einstein model requires the frequencies at the Gamma point for the equilibrium geometry, which are included using the **freqg0** keyword, which associates its input to a phase using its string or integer label (in this case, 'debeins' or 3). MgO has 3 optic branches, so there are 3 frequencies at Gamma to be input, in cm⁻¹, as indicated in the **units** section of the phase.

A preview of the results can be plotted running gnuplot on the 'all_p.gnu' script. For instance, the result for the thermal pressure is:



As the MgO is a closely packed material with not many optic branches the result of the Debye models is quite reasonable, provided a good evolution of the Debye temperature with pressure is used. Thus, Debye-Grüneisen results are very close to QHA while the simple Debye model and Debye-Einstein are farther from the correct quasi-harmonic result. The inclusion of the frequencies at Gamma shifts the thermal pressure towards the QHA result with a small step.

4.4 4.4 Empirical corrections of the energy

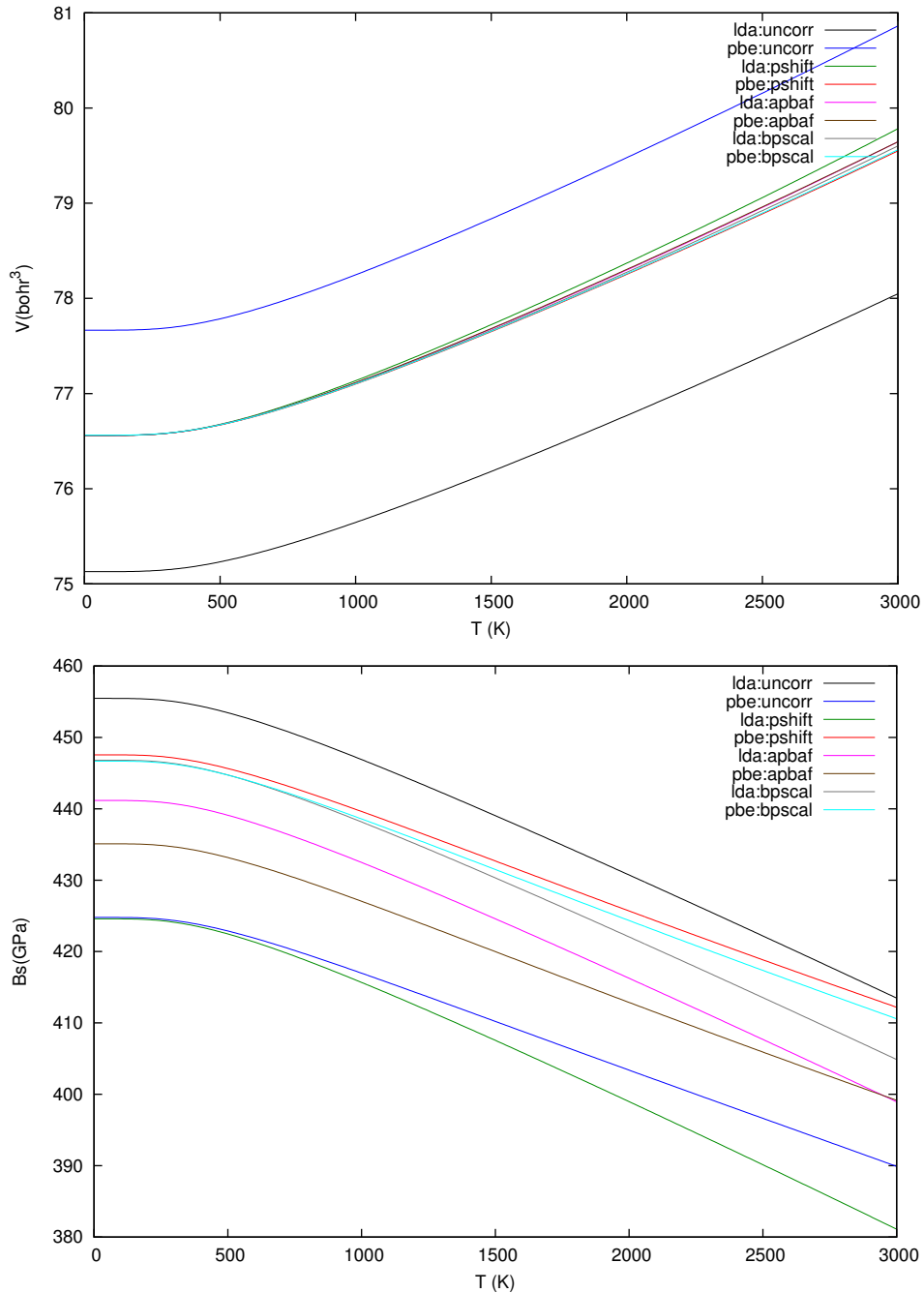
The thermodynamic properties resulting from a density functional calculation most of the time depend dramatically on the exchange-correlation density functional employed, which is the main source of error contained in their calculation provided a good temperature model is used. In order to correct these discrepancies, empirical energy corrections (EEC) have been proposed.

The EEC are corrections on the static energy curve that force the model to meet experimental data. In the current implementation one datum (the experimental equilibrium volume at ambient conditions) or maybe two (the experimental bulk modulus at ambient conditions) are used. Several EECs have been implemented, which are described in full in the reference section and in xxxx. These are the 'pshift' correction (shift the energy with a constant pressure), the 'apbaf' (adding a constant times $1/V$) and the 'bpscal' (using the observation that $p/B0$ versus $v/v0$ is similar for all functionals and experimental data).

In the next example, the three EECs and the unscaled results are calculated for the diamond:

```
1 # Diamond
2 # Bulk modulus: 446 (Ocellli2004)
3 # V0 : 3.4170 cm3/mol (/ 6.02214179e23 * 2 * 1e24 / .52917720859^3) =
4 # = 76.58092 (Ocellli2004)
5 set notrans
6 mm 24.0214
7 vfree 2
8 pressure 0
9 temperature 0 10 3000
10
11 phase LDA:uncorr file ../dat/c_lda/c.res tmodel qha \
12 units energy ry freq cm-1 \
13 prefix ../dat/c_lda/
14 phase PBE:uncorr file ../dat/c_pbe/c.res tmodel qha \
15 units energy ry freq cm-1 \
16 prefix ../dat/c_pbe/
17 phase LDA:pshift file ../dat/c_lda/c.res tmodel qha \
18 units energy ry freq cm-1 \
19 prefix ../dat/c_lda/ \
20 eec pshift 76.58092
21 phase PBE:pshift file ../dat/c_pbe/c.res tmodel qha \
22 units energy ry freq cm-1 \
23 prefix ../dat/c_pbe/ \
24 eec pshift 76.58092
25 phase LDA:apbaf file ../dat/c_lda/c.res tmodel qha \
26 units energy ry freq cm-1 \
27 prefix ../dat/c_lda/ \
28 eec apbaf 76.58092
29 phase PBE:apbaf file ../dat/c_pbe/c.res tmodel qha \
30 units energy ry freq cm-1 \
31 prefix ../dat/c_pbe/ \
32 eec apbaf 76.58092
33 phase LDA:bpscal file ../dat/c_lda/c.res tmodel qha \
34 units energy ry freq cm-1 \
35 prefix ../dat/c_lda/ \
36 eec bpscal 76.58092 446
37 phase PBE:bpscal file ../dat/c_pbe/c.res tmodel qha \
38 units energy ry freq cm-1 \
39 prefix ../dat/c_pbe/ \
40 eec bpscal 76.58092 446
41 end
```

We have used the full QHA model for temperature effects and applied the correction using the experimental volume of the primitive cell (76.58092 bohr³) and the experimental bulk modulus (446) at room temperature. Running the '_all_t.gnu' script file with gnuplot gives:



The first graph is a plot of the calculated volume with temperature. Observe the enormous gap between uncorrected LDA and PBE results, which is reduced to a very close match by energy corrections (and which agree very well with experimental data). All the EECs seem to perform very well for the calculation of $V(T)$, with pshift perhaps being the farthest from the others. The differences in the volume representation are widened when one turns to the calculation of the adiabatic bulk modulus. Again, there is an enormous gap in the uncorrected results. As the bulk modulus is a property that depends mostly on the derivatives of the energy, and not energy itself, the EECs performance varies according to their definition, but all of them seem to improve the result. The best EEC seems to be 'bpascal' which corrects the data to enforce the experimental bulk modulus. The LDA and PBE results corrected with bpascal match closely for lower temperatures and deviate at higher temperatures. The difference between them, though, is still far lower than for the uncorrected gap.

4.5 4.5 The electronic contribution to the free energy

In addition to the vibrational, metals possess electronic degrees of freedom that contribute to the free energy and the heat capacity. These arise from their band structure: the electrons are free to roam the metal because there are empty states at arbitrarily small energies above the Fermi level. Usually, this contribution to the free energy is negligible compared to the vibrational. However, due to its simplicity, gibbs2 provides a means for including it using two different models: the Sommerfeld model of free and independent electrons (**sommerfeld**) and a model that reads the coefficients of a polynomial fitted to results of finite temperature DFT calculations (see Mermin, 1965).

We can illustrate this with the following simple example, where we calculate the properties of fcc aluminium:

```

1 # Aluminium
2 set notrans
3 mm 26.981538
4 vfree 1
5 pressure 0
6 temperature -1
7 volume input
8
9 phase Al:free file ../dat/al_lda/al.res tmodel qha phfield 12 \
10     units energy ry freq cm-1 \
11     prefix ../dat/al_lda/ eec bpscal 112.04 72.7 \
12     elec sommerfeld free nelec 3
13 phase Al:somm_nef file ../dat/al_lda/al.res tmodel qha phfield 12 \
14     units energy ry freq cm-1 edos ev \
15     prefix ../dat/al_lda/ eec bpscal 112.04 72.7 \
16     elec sommerfeld nelec 3
17 phase Al:pol4 file ../dat/al_lda/al.res tmodel qha phfield 12 \
18     units energy ry freq cm-1 \
19     prefix ../dat/al_lda/ eec bpscal 112.04 72.7 \
20     elec pol4 4
21 end

```

The primitive cell of fcc aluminium contains one atom, with molecular weight 26.981538 amu. The pressure list is composed only of the zero pressure point and the temperature is room temperature. It is possible direct gibbs2 to calculate the properties at given temperatures on a list of volumes instead of pressures with the **volume** keyword. The **volume input** order instructs gibbs2 to calculate the properties at the input grid of volumes.

Three phases are included in the input, each representing a different electronic model. The first phase corresponds to the free electron model, which is activated using the **elec sommerfeld free** order. The model requires the number of conduction electrons per primitive cell, 3 in this case. The energy is corrected to yield the experimental volume (112.04 bohr³) and bulk modulus (72.7 GPa) of aluminium with an EEC as described above, which can be used with different temperature and electronic models. The second phase corresponds to the free electron model but where the population at the Fermi level, $N(E_f)$, is read from the input file. This is achieved by omitting the **free** keyword. In the third phase, the coefficients of the fourth degree polynomial that fits F_{el} (the electronic Helmholtz free energy) and $-T*S_{el}$ with respect to temperature are read. This option requires the usage of an external fitting script to find the coefficients.

The input datafile is always the same (al.res), and its structure is (the backslashes are not contained in the original input file, the lines have been wrapped for clarity):

```

1 35.200000000000003 -11.447724709999999 0.203700 2.0620541311e-09 \
2 -1.8439425798e-10 7.4740480714e-15 -5.5458673605e-18 \
3 -9.2793212377e-10 -3.5154688019e-10 -9.1949143703e-15 \
4 -4.2538048412e-18 012_/012_.phdos
5 ...

```

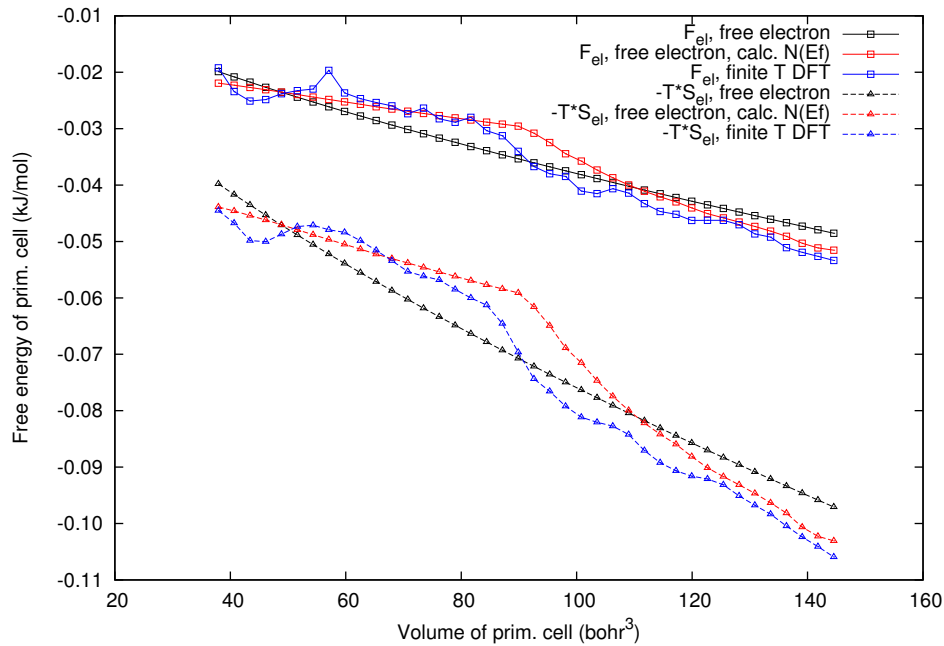
As always, the first field is the primitive cell volume and the second is the energy. Then, $N(E_f)$ in 1/eV (see **units edos** in the ing file above) and the 8 polynomial coefficients. The first 4 coefficients (fields 4 to 7) fit F_{el} with respect to T (T coefficient is field 4, T^2 is field 5,... there is no independent term because $F_{el}(0) = 0$). The next 4 coefficients (fields 8 to 11) fit $-T*S_{el}$ to temperature. The units of both F_{el} and $-T*S_{el}$ are the units set by the **units energy** keyword, that is, Ry. The last field points to the phonon density of states file for that volume.

The ordering of the fields must be treated carefully. By default, the interpretation sequence is: volume, energy, Debye temperature, $N(E_f)$, coefficients of F_{el} , coefficients of $-T*S_{el}$ and phonon file (DOS or frequencies). Depending on the phase, some of these fields are read and some are not. This order is, however, not compulsory, and the order of each of the fields can be modified by the appropriate keyword. In the first phase, the free electron model and QHA are used. The free electron model does not need any of the above variables, but QHA does require the vibrational input file. Because the default field for phdos1.s is 3 (only volume and energy are read, and come before the phDOS field), it is necessary to indicate to gibbs2 that the file is actually located in field 12 in the input file. This is achieved using the **phfield** keyword after **qha**. The same applies to the other two phases.

In the second phase, the free electron model with $N(E_f)$ input is used. The field from which the $N(E_f)$ is read can be changed using an integer right after the **sommerfeld** keyword. If no integer is found, then the default field is used. In the al.res input file, $N(E_f)$ appears on the third field, right after the energy. As the Debye temperature is not read in QHA, 3 is the default field for $N(E_f)$, and it is not necessary to write explicitly the field where $N(E_f)$ lives.

In the third phase, the 8 polynomial coefficients and the phDOS file are read. The default field for the first polynomial coefficient is 3, but the position it occupies in the file is 4, so it is necessary to write it explicitly in the input line, after **pol4**. Note that the 8 coefficients must always appear in sequence, with the correct order.

The following plot represents the results of the above input:



It represents the electronic Helmholtz free energy (the upper bunch of curves) and $-T*S_{el}$ against the input volume grid. Al is the prototype of a free electron metal, which is clearly confirmed by the aparametric free electron model rendering free energies extremely close to those calculated by ab initio programs. Note the noise in the calculated F_{el} and TS_{el} and even in the Sommerfeld model using the external $N(E_f)$. One kelvin is more or less π microHartree, so it is clear that the energy correction a moderate temperature (say, 500 K) has on the static energy is quite small. Therefore, the accuracy of this correction is extremely dependent on the calculation parameters and, in particular, on the k-point grid. The results shown above are calculated using a 52x52x52 Monkhorst-Pack grid. A final note: the magnitude of F_{el} is small (some %) compared to F_{vib} , so the thermodynamic properties (equilibrium volumes, bulk moduli, etc.) are barely affected, except at extremely low temperatures.

4.6 Phase transitions

Contrarily to the old gibbs, gibbs2 can actually calculate stability diagrams and transition pressures between different phases of the same solid by calculating the Gibbs free energy for each of them and then determining the most stable phase in the input at given pressure and temperature conditions.

In the next example, we have calculated the transition pressures with temperature of MgO, from the very stable B1 \rightarrow B2 phase. It is clear from the literature that this transition occurs around 500 GPa, but it has not been experimentally observed yet (jan. 2011).

An important note: when comparing phases with different number of atoms in the cell it is necessary to scale the extensive quantities given in input (volume, energy,...). See the **vf** and **z** keywords for further information.

The input file for the MgO B1 \rightarrow B2 is:

```
1 mm 40.3044
2 vfree 2
3 pressure 0 25 600
4 temperature 0 20 1000
5
6 phase B1 file ../dat/mgo_pbe/mgo.res tmodel qha \
7     units energy ry freq cm-1 \
8     prefix ../dat/mgo_pbe/ \
9     eec bpscal 126.025 161.3
10 phase B2 file ../dat/mgob2_pbe/mgob2.res tmodel qha \
11     units energy ry freq cm-1 \
12     prefix ../dat/mgob2_pbe/ \
13     eec use 1
```

This time, each **phase** keyword corresponds to a physical phase, and the two data files are different. Note that the energy correction is applied as usual to the first phase, B1, which is the stable phase at ambient conditions. There is absolutely no experimental information about the B2 phase, so we opt here to use the same correction coefficients as in the B1 phase (number 1), with **eec use 1**.

The gibbs2 output contains the static transition pressures:

```
1 * Static transition pressures (linear interpolation)
2 #           Pressure range (GPa)           Stable phase
3           0.0000 -->           503.1471           b1
4           503.1471 -->           600.0000           b2
```

The static transition pressure is 503.1 GPa, more or less the same result as other authors found for this transition. In addition, gibbs2 generates a number of auxiliary files:

- 06_phases_dH.gnu, 06_phases_dH.aux : this files generate a plot of the difference in static enthalpy between all the phases and the first phase in the input. In this case, the plot is:
- 06_phases.tpstab : a file containing the identity, Gibbs free energy, volume and adiabatic bulk modulus of the stable phase at each pressure and temperature data point.
- 06_phases.dgtp : difference in Gibbs free energy between all the calculated phases and the first, for all the temperature and pressure data points.
- 06_phases.ptrans, 06_phases_ptrans.gnu : a plot of the evolution of the transition pressure with temperature. A slight modification of the gnu file yields:

```
1 #           T (K)           b1           b2
2           0.0000           0.0000 ->           492.1215 ->           600.0000
3           20.0000           0.0000 ->           492.1215 ->           600.0000
4           ...
5
6 Each '->' symbol corresponds to a stability domain, *not* a
7 transition. For instance, at 20 K, the B1 phase is stable from 0 to
8 492.1 GPa. The B2 phase is stable from the latter pressure to 600
9 GPa at least. 600 GPa is the highest pressure explored in the input.
```

5 5 Keyword reference

5.1 5.1 General purpose keywords

A full list of the gibbs2 input syntax follows. Integer, real and string variables are denoted by i, r and s suffixes respectively. Optional arguments are enclosed in brackets. Alternatives are grouped by curly braces and separated using a bar (|).

- **# This is a comment**

Comment lines begin with a # character. Lines are continued with a backslash (\) character

- **title title.s**

The title of the run.

- **nat nat.i, vfree nat.i**

Number of atoms per (primitive) unit cell or per unit formula (both keywords are equivalent). Any multiple of the basic unit formula can be used, provided some conditions are met. These conditions are:

- In the **static** temperature model, **nat.i** is not used whatsoever, so this keyword can be omitted. However, volumes and energies must be consistent with each other.

- In **debye** and **debye_gruneisen**, **nat.i** and **mm** must refer to the same basic unit of repetition. In addition, **nat.i** times the **Z** of each phase (see the PHASE keyword below) must be the number of atoms for which energies and volumes are calculated. If the model is **debye_input**, then the Debye temperatures must be normalized to **nat.i** atoms.
- In **debye_einstein**, the same as in **debye** applies. The number of frequencies read is $3 \times \text{nat.i} \times Z - 3$, where **Z** depends on the phase.
- In **qha**, the same as in **debye** applies. Additionally, the phonon DOS must be normalized to $3 \times \text{nat.i} \times Z$.
- In **qha_espresso**, the same as in **debye** applies. The number of frequencies read is $3 \times \text{nat.i} \times Z$ for each reciprocal space point.

Note that, with independence of the temperature model used, the phonon density of states are renormalized to **nat.i** so the extensive quantities in the output are per **nat.i** atoms, regardless of the **Z** of each phase. This allows free energy comparisons.

To simplify, unless there is a good reason (e.g. two different phases with different number of non-equivalent atoms and hence frequencies per k-point), it is easier to input quantities per primitive cell.

- **mm mm.r**

Mass (amu) per unit containing **nat.i** (defined using **vfree**) atoms.

- **nelectrons nelec.i**

Total number of electrons in the primitive unit cell. This value is used only if any of the phases is fit using Holzapfel's AP2. The value of this variable implies nothing to the value of the **nelec** keyword in the **phase** keyword.

- **einf einf.r**

Energy at $V=\text{Inf}$. Used only for the scaling of the extensive energetic properties (default 0).

- **pressure** keyword

```

1 pressure pini.r pstep.r pend.r
2 pressure pstep.r
3 pressure npres.i
4 pressure
5   p1.r p2.r p3.r ...
6   p4.r ...
7 endpressure
8 pressure 0

```

The list of pressures where the thermodynamic properties are calculated. **pini.r**, **pstep.r** and **pend.r** determine a pressure range: from **pini.r** up to **pend.r** in steps of **pstep.r**. If only **pstep.r** is given, **pini.r** is assumed to be 0 and **pend.r** is the highest pressure possible from the input data. If **npres.i** is given, use **npres.i** pressures in that same range. Note that, if **pstep.r** is desired, then it must be clearly a real number to differentiate it from a **npres.i** input. A list of pressures can be given in the form of the environment **pressure ... endpressure**. **pressure 0** uses only the zero pressure.

Default: 100 points from 0 to $\max(p, 500)$

- **volume** keyword

```

1 volume vini.r vstep.r vend.r
2 volume vstep.r
3 volume nvols.i
4 volume
5   v1.r v2.r v3.r ...
6   v4.r ...
7 endvolume
8 volume input

```

In addition to using a pressure list, properties can be calculated at states given temperature and **volume**. The **volume** keyword expresses the list of volumes where properties are calculated. The syntax is equivalent to that of **pressure**. Using **volume input**, the volumes of the input grid are used.

Default: no volumes.

- **temperature** keyword

```

1 temperature tini.r tstep.r tend.r
2 temperature tstep.r
3 temperature ntemp.i
4 temperature
5   t1.r t2.r t3.r ...
6   t4.r ...
7 endtemperature
8 temperature 0
9 temperature -1

```

Same as above: the **temperature** keyword builds the list of temperatures where the thermodynamic properties are calculated. If **temperature** 0 is used, the list only contains the 0 K value. **temperature** -1 makes the room temperature (298.15 K) the only element in the list.

Default: 100 points from 0 to 1.5max(td).

- **freqg0** keyword

```

1 freqg0 {name.s|num.i} [file file.s]
2   # comment
3   freq1.r freq2.r ...
4   ...
5 endfreqg0

```

Optical Frequencies at gamma for the phase with string identifier name.s or integer identifier num.i. The latter corresponds to the order in which phases appear in the input. The frequencies are freq1.r, freq2.r,... and there must be exactly 3*nat.i-3 of them (nat.i is defined using **vfree**). These frequencies are used only in the Debye-Einstein model (see below). Optionally, the information in the environment can be input from an external file file.s.

- **interpolate** keyword

```

1 interpolate input [static]
2 interpolate
3   [P]
4   p1 p2 ..
5   p3 ..
6   V
7   v1 v2 ..
8   PT
9   p1 t1 p2 t2 ...
10 endinterpolate

```

In the definition of a phase, it is possible to define satellite data (e.g. internal coordinates, cell parameters,...) using the phase keyword **interpolate**. These data are meant to be linear interpolated at some chosen volumes, which are input with the **interpolate** keyword. There are three possible modes in which the input in the environment is interpreted, that are activated by 'p', 'v' or 'pt':

- **v**: give the volume directly.
- **p**: use a static pressure and interpolate to the corresponding static volume.
- **pt**: interpolate to the equilibrium volume at the given pressures and temperatures. The input is read in pairs: first a pressure then a temperature, iterating until the input is exhausted.

When the **input** keyword is used, the input pressures and temperatures are used. With **static**, only the input pressures under static conditions are used. If 'pt' values are used with the static temperature model, they are ignored.

- **activate {all|v1.i v2.i v3.i...}** In the qhafull temperature model, gibbs2 automatically deactivates a volume when the phonon density of states in input contains negative frequencies. However, because of numerical errors in the DFPT calculation and posterior Fourier interpolation, it is possible to have a small region of negative frequencies. In such a case, it is possible to activate manually the use of those volumes in the dynamic calculation with **ACTIVATE**. With the **ALL** keyword, all the volumes become active. Alternatively, the user can input the volume integer identifier (the position of the volume in the input grid). This identifier is written to the output when a volume is deactivated because of negative frequencies.

- **printfreq|printfreqs** Print the calculated frequencies in the Debye-Einstein model for all static volumes in input. Writes the .gammafreq file.
- **eoutput [vini.r vstep.r vend.r]** Print the static energy of each phase to external files (extension edat). Without any options, the volume grid in input is used. This option can be used to print the corrected static energies when using EECs.
Also, a new volume grid can be chosen by indicating an initial (vini.r), final (vend.r) and volume step (vstep.r). This is useful when extrapolating the input static energy. Please note that it is important to use a low-order EOS (like, for instance, FIT STRAIN BM 3) to extrapolate. The default averages of strain polynomials behave badly on extrapolation.
- **drhouse**
Detects problems (noise, outliers,...) in the input data using a bootstrap technique: random subsets of data are fitted to polynomials of varying degree. The statistics of this process allows gibbs2 to assess the quality of the data.
- **end**
Ends the run. This keyword is not necessary for a correct termination of gibbs2.

5.2 The phase keyword

```

1 phase name.s \
2   [file file.s [u|using a:b[:c]]] \
3   [Z z.r] \
4   [poisson sigma.r] \
5   [laue laue.s] \
6   [fit {polygibbs|bm2|bm3|bm4|pt2|pt3|pt4|pt5|murn|antons|vinet|ap2|
7       strain {eulerian|bm|natural|pt|lagrangian|lagr|
8           infinitesimal|inf|quotient|x1|x3|xinv3|x3inv|v} [order.i|0]]}\
9   [reg {lad|lsq}] \
10  [fix i1.i v1.r i2.i v2.r ...] \
11  [tmodel {static|debye_input|debye_poisson_input|debye|debye_einstein|
12      debye_gruneisen {slater|dm|vz|mfv|a.r b.r}|
13      {qhafull|qha} [phfield ifield.i] [dosfield i1.i i2.i]|
14      qha_espresso [phfield ifield.i]] \
15  [prefix prefix.s] \
16  [elec sommerfeld free
17      sommerfeld [icol.i]
18      pol4 [icoll.i]] \
19  [nelec nelec.i] \
20  [eec nocal|pshift vexp.r|bpscal vexp.r bexp.r|apbaf vexp.r|
21      use phase.i] \
22  [eec_p pext.r] [eec_t text.r]
23  [eshift eshift.r]
24  [pvdata] \
25  [units {volume {bohr|bohr3|bohr^3|ang|ang3|ang^3}}
26      {energy {hy|hartree|ha|ev|evolt|electronvolt|ry|rydberg}}
27      {pressure {au|a.u.|gpa}}
28      {{freq|frequency} {hartree|hy|ha|cm-1|cm^-1|cm_1|thz}}
29      {edos {hy|hartree|ha|ev|evolt|electronvolt|ry|rydberg}}] \
30  [interpolate f1.i [f2.i ...]]
31  [fstep step.r]
32  # comment
33  v1.r e1.r [td1.r nef1.r f1.r f2.r f3.r f4.r ts1.r ts2.r ts3.r ts4.r phdos1.s
34      int1.r int2.r ...]
35  ...
36 endphase

```

Adds a phase to be analyzed. The phase can be a physical phase, that is, it corresponds to a real phase of the solid under study, or a different representation of the same dataset: different fits, thermal models, etc. Each phase is associated with two identifiers: a string (name.s, case insensitive) that labels the phase in the output and plots; and a number, assigned by the order of appearance of phases in the input, starting from 1. The static and dynamic properties of all the phases are calculated by gibbs2.

The phase keyword has a number of options, that are written right after the phase identifier (the order is not important). All the keywords are optional: if none are input, reasonable default values are assumed. A correct description of a phase requires a minimum of external data: the static energy curve. This means that, using an external program the user is required to calculate the value of the nuclear potential E on a grid of volumes, most likely encompassing the equilibrium volume of the solid (although, actually, the existence of a minimum is not necessary for gibbs2 to work with a dataset). The energy-volume data is input after the **phase** keyword in a series of records containing at least two fields: the volume and the energy, in that order. Depending on the thermal model, electronic model,... a number of additional fields can be read. The data input ends with an **endphase** keyword. We will describe below the exact sequence in which these fields are interpreted.

This input method is sometimes clumsy, as it requires the user to copy the results of the ab initio calculation into the input file for gibbs2. It usually is more comfortable to read an external file containing precisely the same input in the **phase** environment. This is easy using the **file** keyword, with the name of the external file being file.s. If the **file** keyword is used, no records after **phase** or **endphase** are needed. The lines of this file that start with a '#' character are comments, and ignored by the parser. By default, the first field of each record is assigned to volume and the second is the energy. This behavior can be modified with the **using** (or **u** for short) keyword. For instance:

```
1 phase mgo file mgo.dat u 4:5:8
```

reads the file mgo.dat: the fourth field is interpreted as the cell volume and the fifth is the energy. The optional third field identifier is assigned to the Debye temperature or the Poisson ratio. It is read only when the `debye_input` or the `debye_poisson_input` temperature models are used, respectively. It is ignored otherwise.

A description of the rest of keywords optional to **phase** follows:

- **z** z.r

The **z** keyword is useful in the context of comparing phases of the same solid with different number of atoms in the primitive cell. To make all the phases consistent, it is necessary that the volumes, energies, frequencies and phonon density of states in the input are given per `nat.i` times `z.r`. The `nat.i` value (set with the **vfree** keyword) is the same for all the phases, and `z.r` must be set to the value that fulfills the condition above for each phase.

Internally, the effect of using **z** is that volumes and energies are divided by `z.r`, and the `phDOS` are renormalized. This makes the extensive quantities of all the phases be referred to the same amount of matter: the number of atoms set by **vfree**. Of course, **vfree** is arbitrary, but the sensible choice is to make it equal to the greatest common divisor of the number of atoms in the cell of all the phases to keep `z.r` an integer.

If the `z.r` of every phase are correctly set, the intensive quantities in the output should be independent of the `nat.i` and `z.r` used. The extensive quantities are reported per `nat.i` atoms.

Default: 1.

- **poisson** sigma.r

The Poisson ratio of the phase. It is used in the calculation of the Debye temperature when using Debye thermal models.

Default: 1/4 (the Poisson ratio of a Cauchy solid).

- **laue** laue.s

The Laue group of the crystal. It can take the values:

- -1, ci
- 2/m, c2h
- mmm, d2h
- -3, c3i
- 4/m, c4h
- 4/mmm, d4h
- 6/m, c6h
- 6/mmm, d6h
- m-3, th
- m-3m, oh

The Laue group is used in the `qha_espresso` thermal model. The vibrational frequencies on a q-point grid are read, but only for a irreducible symmetry subset of the full q-point grid. Therefore, the symmetry operations of the crystal are needed in order to assign weights for each q-point. These operations are internally codified: the user is only required to provide the Laue group.

Default: `laue.s` is required **tmodel qha_espresso** is used. It is ignored otherwise.

- **fit** keyword

```
1 fit {polygibbs | bm2 | bm3 | bm4 | pt2 | pt3 | pt4 | pt5 | murn | antons | vinet | ap2 |
2      strain {eulerian | bm | natural | pt | lagrangian | lagr |
3      infinitesimal | inf | quotient | x1 | x3 | xinv3 | x3inv | v} [order.i | 0]
```

The **fit** keyword chooses the equation of state (EOS) used for the fits to the energy-volume and free energy-volume data of the phase. A large number of EOS are available, but they can be grouped by how their parameters are found. If the EOS are expressed as a polynomial in some strain, that is, a given function of volume expressing the deviation of the geometry from a given reference, then it is possible to carry out the fit using a linear polynomial fitting algorithm. In any other case, non-linear minimization techniques need to be used. In **gibbs2**, the **slatec** library is used for the fitting of polynomials (unless **set pweigh_mode** changes this behaviour) and **minpack** for the non-linear fits. The latter uses a Levenberg-Marquardt algorithm, for which the convergence to the desired solution is not guaranteed. Therefore, it is in general a good idea to use linear fits if possible. Additionally, linear fits allow the user to fit strain polynomials to degrees much higher than non-linear fits: for instance, Birch-Murnaghan 5+th order.

Linear fits (fits of polynomials of some strain) are requested by usgin the **strain** keyword. The word following it must be the type of strain to be used, that can be one of:

- **eulerian** or **bm**, Birch-Murnaghan or eulerian strain, $f = ((V/V_0)^{-2/3} - 1)/2$
- **natural** or **pt**, Poirier-Tarantola or natural strain, $f = \log(V/V_0)/3$
- **lagrangian**, lagrangian strain, $f = ((V/V_0)^{2/3} - 1)/2$
- **infinitesimal**, infinitesimal strain, $f = -(V/V_0)^{-1/3} + 1$
- **quotient** or **x1**, compression, $f = V/V_0$
- **x3**, $f = (V/V_0)^{1/3}$
- **xinv3** or **x3inv**, $f = (V/V_0)^{-1/3}$
- **v**, $f = V$

Following the strain, the user must indicate the order of the polynomial to be fit. Thus, for instance,

```
1 phase mgo ... fit strain bm 4
```

fits a 4th order Birch-Murnaghan equation of state to the energy, which is equivalent to a 4th degree polynomial of the energy in the **bm** strain. If 0 is used as the degree of the polynomial, then an average is used, as described in the companion articles of this source. In short, a number of polynomials of degree varying from **mparmin** to **mpar** (see the corresponding set options) are fitted to the data. Each polynomial is assigned a weight, which is larger the better it fits the data, but smaller the more parameters the polynomial uses. The weights are used to find an average polynomial of degree **mpar**, which is used as the final fitting function. The statistics of the polynomials also provide a measure of the quality of the data and of the reliability of the fit. Note that **fit strain bm 0** is the default behavior for all the phases if no **fit** keyword is found. We have found this technique to be extremely robust.

In addition, the old fitting technique from **gibbs1** can be used with **polygibbs**. This option is provided for compatibility reasons, but we consider it superseded by the above. The new version does *not* remove points from the ends of the volume grid.

Sometimes it is interesting to use a low-order equation of state, most likely to extrapolate the behavior of a phase outside of the calculated volume grid. If the desired EOS has no expression in a polynomial expansino of some strain, then non-linear fits need to be carried out. The EOS implemented in **gibbs2** for non-linear fitting are:

- **bm2**: Birch-Murnaghan, second order.
- **bm3**: Birch-Murnaghan, third order.
- **bm4**: Birch-Murnaghan, fourth order.
- **pt2**: Poirier-Tarantola, second order.
- **pt3**: Poirier-Tarantola, third order.
- **pt4**: Poirier-Tarantola, fourth order.
- **pt5**: Poirier-Tarantola, fifth order.
- **murn**: Murnaghan.
- **antons**: Anton-Schmidt.
- **vinet**: Vinet.
- **ap2**: Holzapfel's AP2.

The use of **bm**x and **pt**x are strongly discouraged: use **fit strain bm** x and **fit strain pt** x instead. When the non-linear fits converge, the results are completely equivalent.

Default: **fit strain bm** 0.

- **reg** {ladlsq}

Choose the regression technique for the fits: least-squares (**lsq**, default) or least absolute deviation (**lad**). The former minimizes $\sum(|y_i - f(x_i)|^2)$ while the latter minimizes $\sum(|y_i - f(x_i)|)$. The latter is sometimes used as a robust fitting technique, because it is less sensitive than least-squares to noise in the data. **lad** can only be used together with non linear fits (*not* with **fit strain x y** or **fit polygibbs**) .

Default: **lsq**.

- **fix** i1.i v1.r i2.i v2.r ...

Fixes some of the EOS parameters to user-defined values. This keyword only applies to non-polynomials (i.e. not **fit strain** or **fit polygibbs**) and to fits of static data: the constraints are not honored when fitting free energy vs. volume curves. The i1.i identifier is the parameter to be fixed, and can be V0 (2), B0 (3), B0' (4), B0'' (5), B0''' (6). The v1.r is the value with which the parameter is fixed. Several parameters can be fixed by adding more i.i v.r pairs to the list.

Default: no parameters fixed.

- **tmodel** keyword

```
1 tmodel {static|debye_input|debye_poisson_input|debye|debye_einstein|
2         debye_gruneisen {slater|dm|vz|mfv|a.r b.r}|
3         {qhafull|qha} [phfield ifield.i] [dosfield i1.i i2.i]|
4         qha_espresso [phfield ifield.i]}
```

Sets the temperature model. The keyword meaning is:

- **static**: do not calculate thermal properties of this phase.
- **debye**: Debye model, where the Debye temperature at each volume is calculate from the bulk modulus and the Poisson ratio, as in Slater's 'Introduction to Chemical Physics'. The Poisson ratio is assumed to be constant and equal to the value given in the **poisson** keyword, or 1/4 by default.
- **debye_input**: Debye model, the Debye temperatures are read from input. Each volume-energy record must contain an additional field, corresponding to the Debye tempearture at that volume.
- **debye_input**: Debye model, the Poisson ratio at each volume are read from input. Each volume-energy record must contain an additional field, corresponding to the Poisson ratio at that volume.
- **debye_einstein**: Debye model for the acoustic branches and $3n-3$ Dirac deltas representing the optic part of the phonon spectrum. The Debye temperature is calculated as in the **debye** model, but normalized to 3. The frequencies at gamma for the equilibrium or experimental geometry are input with the **freqg0** keyword, and they are used to represent the optic part, using an approximate volume dependence.
- **debye_gruneisen**: Debye model, where the Debye temperature is calculated for the equilibrium volume as in **debye**, but the volume evolution of this temperature is controlled by an approximate gamma, of the form $a + b \cdot B'$, where B' is the pressure derivative of the static bulk modulus. There are 4 keywords that represent predefined values of these parameters, but the user has the freedom to choose a.r and b.r by indicating so in the input. The predefined choices are:
 - * **slater** : Slater gamma, $a = -1/6$ and $b = 1/2$. This is equivalent to the **debye** model
 - * **dm** : Dugdale-McDonald gamma, $a = -1/2$ and $b = 1/2$.
 - * **vz** : Vaschenko-Zubarev gamma, $a = -5/6$, $b = 1/2$.
 - * **mfv** : Mean free volume gamma, $a = -0.95$, $b = 1/2$.
- **qhafull** or **qha**: full quasiharmonic approximation. The calculated phonon density of states is found at each volume of the input grid and fed into gibbs2. Each record of the **phase** environment thus requires an additional field, pointing to the file that contains the phonon density of states. By default, the field read is right after the static energy (see below for a description of the interpretation sequence of the environment records, though), but this behavior can be modified with the **phfield** keyword. This keyword is the equivalent of the **using** keyword for the volume and energy. Fields 1 and 2 are read from the files that contain the vibrational density of states: the first is the frequency (default units Ha) and the second is the density of states (default units 1/Ha), but which fields are read can be controlled using the **dosfield** keyword. The **prefix** keyword below is relevant to the determination of the path of these files.

- **qha_espresso: full quasiharmonic approximation.** Instead of reading the vibrational density of states, a file containing a description of a sampling of the 1BZ and the frequencies at the q-points is read. The format is that of the *freq* file of Quantum ESPRESSO, hence the name of the keyword. Note that this model requires the Laue group of the crystal to find the multiplicity of each q-point, in order to assign the weights. As in the previous model, the name of the file that contains the frequencies at each volume is read from the field right after the static energy, or the field *ifield* if **phfield** is used. Again, the **prefix** keyword is very relevant.

Some notes: the points where there exist negative frequencies (**qha_espresso**) or the vibrational DOS presents a non-null integral in the negative frequency region (**qha**) or where the second derivative of the static energy is negative (Debye-like models, past the spinodal point) are deactivated for thermal calculations in their respective models. This means that they still enter the static fits, but a reduced volume grid where these points have been removed is used for dynamic calculations and free energy fits.

In the regions where negative frequencies start to appear, it is best to use the phonon DOS (**qha**) instead of the discrete frequencies on a q-point grid (**qha_espresso**). In fact, in the examples we have examined, there is little gain in using **qha_espresso** instead of **qha**.

Default: **debye**.

- **prefix** prefix.s

Prefix for finding the files that contain the vibrational density of states or frequencies. For instance, let us consider an input phase such as:

```
1 phase tmodel qhafull
2   81.8883583665837 -73.5171659350000 001/001.dos
3   86.0358791612784 -73.5360133400000 002/002.dos
4 endphase
```

Without any **prefix** keyword, **gibbs2** expects that the files containing the phonon density of states are located in *./001/001.dos*, *./002/002.dos*,... where *./* is the place where the directory is launched, that is usually almost always the one containing the *ing* file. However, sometimes it is interesting to read the energy-volume and phonon density of states from somewhere else. For instance, if the density of states were located in a directory called */home/user/calc*,

```
1 phase tmodel qhafull prefix /home/user/calc
2   81.8883583665837 -73.5171659350000 001/001.dos
3   86.0358791612784 -73.5360133400000 002/002.dos
4 endphase
```

then **gibbs2** looks for the phonon density of states in */home/user/calc/00x/00x.dos*. It is also possible to include relative paths, e.g., **prefix ../calc**.

Default: *'./'*.

- **elec** {sommerfeld [freelicol.il]|pol4 [icol.i]}

Activate the calculation of the electronic contribution to the properties of the solid, only relevant for metals. There are currently two models implemented. With the **sommerfeld** keyword, the Sommerfeld model of free and independent electrons is used. Furthermore, if **free** is used, then the free electron Fermi energy and density of states are used. If *icol.i* is used instead, the occupation (total, up+down) at Fermi level is read from an additional column in the input file (see the end of this subsection for the interpretation sequence). With **pol4**, it is possible to input the result of finite temperature density functional calculations. Starting from *icol.i* (the default value depends on the temperature model,... see the interpretation sequence below), 8 fields are read. The first 4 fields are the coefficients of a 4th degree polynomial fit to the electronic free energy (*Fel*) with respect to temperature. That is, if the fields are *i1*, *i2*, *i3*, *i4*, $Fel = i4 * T^4 + i3 * T^3 + i2 * T^2 + i1 * T$. The fields 4-8 represent $-T*S_{el} = i5 * T^4 + i6 * T^3 + i7 * T^2 + i8 * T$. Usually, one fits the total free energy with respect to volume to find the first four and the $-T*S$ contribution to find the last four. Both output fields should be available from any ab initio program capable of this kind of calculations. Note that the **sommerfeld** model requires the number of conduction electrons given by **nelec**, as explained next.

The coefficients related to the **POL4** keyword are affected by the value of *Z*, in the same way as volumes and energies. That is, the fitted free energy contributions should correspond to *Z* units.

Default: no electronic contribution.

- **nelec** nelec.i

The number of conduction electrons in the solid. This value is only required if an electronic contribution model is used together with **sommerfeld**. Note that this keyword has nothing to do with the general **nelectrons** keyword, which is used only for the AP2 fits.

Default: 0.

- **eec** {noscallpshift vexp.r|bpscal vexp.r bexp.r|apbaf vexp.r|use phase.i}

Activate the use of an empirical energy correction (EEC) for this phase. The EECs correct the original static energy to fit some experimental data. The experimental data is the volume at zero pressure and room temperature and, optionally, the bulk modulus (in the case of **bpscal**). The expressions of the energy corrections follow:

- **noscal**: do not use any correction.
- **pshift**: $E' = E + \Delta P * V$, with parameter ΔP .
- **bpscal**: $E' = E + Bexp * Vexp / B0 / V0 * (E(V * V0 / Vexp) - E(V0))$, with parameters $Vexp$ and $Bexp$.
- **apbaf**: $E' = E + \alpha / V$, with parameter α .
- **use**: if the phase is not stable at room conditions, the equilibrium volume and bulk modulus are not available. With use, the EEC of a previous phase is copied. This is useful in the context of phase transitions. phase.i is the integer identifier of the source phase.

Default: noscal.

- **eec_p** pext.r

Specifies the external pressure at which the experimental parameters for the empirical energy correction were obtained (in GPa).

Default: zero pressure.

- **eec_t** text.r

Specifies the external temperature at which the experimental parameters for the empirical energy correction were obtained (in K).

Default: room temperature.

- **eshift** eshift.r

Displaces the static energy by eshift.r. The units are the same as in input.

Default: 0.

- **pvdata**

Interprets the input data as $p(V)$ instead of $E(V)$. The appropriate equation of state is fit to the data and then an $E(V)$ curve is generated by integration. The volume grid is somewhat extended in the $V > V0$ region (that is, to negative pressures), because of gibbs2 internal requirements. Relevant **set** options control the number of expansion points generated (**newpts**) and the extent of the expansion (**facexpand**). Interpolation and Debye temperatures from input are not compatible with this order. Also, this order is experimental. Note that the **pvdata** uses the polynomial fit scheme, independently of the **FIT** keyword.

Default: read $E(V)$.

- **units**

```
1 units [ {volume {bohr|bohr3|bohr^3|ang|ang3|ang^3}} ]
2       [ {energy {hy|hartree|ha|ev|evolt|electronvolt|ry|rydberg}} ]
3       [ {pressure {au|a.u.|gpa}} ]
4       [ { {freq|frequency} {hartree|hy|ha|cm-1|cm^-1|cm_1|thz}} ]
5       [ {edos {hy|hartree|ha|ev|evolt|electronvolt|ry|rydberg}} ] \
```

Specifies the units of input data. All data is converted to atomic units internally, so most output is also in atomic units, except where it is explicitly noted. Pressure input units are relevant to the **pvdata** keyword. The frequency units apply to the vibrational density of states, frequencies at Gamma,... The **edos** units apply to the Fermi level read in **edos sommerfeld** with input $N(E_f)$.

Defaults: bohr³ (V), Ha (E), GPa (p), Ha (frequency) and Ha (edos).

- **interpolate** f1.i [f2.i ...]

Define the fields f1.i, f2.i, ... in the input environment as interpolation fields. The interpolation is done at the volumes specified in the external **interpolate** keyword.

Each record in the input environment of a **phase** contains a varying number of information fields, depending on the temperature model, electronic model,... The default interpretation is the following: volume, static energy, Debye temperature, population at the Fermi level, polynomial coefficients for the electronic free energy, polynomial coefficients for -TSel and phonon DOS or frequency file. Therefore, for an input like:

```
1 phase tmodel debye_input elec sommerfeld
```

the input records must be:

```
1 v1 e1 td1 nef1
2 ...
```

The default sequence can be modified by using some of the relevant keywords: **using**, **phfield**,... If that is the case, the fields assigned using those keywords are reserved for the property, and the rest are assigned in order using the default sequence, starting from field number 1.

- **fstep** step.r

When using the qhafull temperature model, it is necessary for gibbs2 to interpolate the phonon DOS at arbitrary volumes. Therefore, all the phDOS read from input are interpolated to the same frequency grid. By default, the step of this grid is the difference in frequencies between the two highest frequencies of the first volume in input. With fstep, it is possible to modify this default behavior.

5.3 Set options

- **set root root.s**

The file root for generated output (root.eos, etc.).

Default: the name of the input file without extension.

- **set pfit_mode {gauss|slatec}**

Use the gibbs1 gauss method to solve the linear system when fitting polynomials (gauss) or dpolft/dpolcf from slatec library (slatec). The former is known to be unstable.

Default: slatec.

- **set pweigh_mode {gibbs1|gibbs2|slatec}**

Use the gibbs1 or gibbs2 method of average of fitting polynomials (gibbs2) or let slatec do it (slatec). The gibbs2 method is strongly recommended.

Default: gibbs2.

- **set mpar mpar.i**

Sets the maximum degree of the weighed polynomial fit to mpar.i. It is important to set this parameter to an adequate value when using datasets with a low number of E(V) points.

Default: 12. The value actually used in the fitting is between 3 and half the E(V) points minus one.

- **set mparmin mparmin.i**

Sets the minimum degree of the weighed polynomial fit to mparmin.i, only in gibbs2 pweigh_mode.

Default: 12.

- **set ndel ndel.i**

The number of external points to be removed from the dataset in polygibbs. Note that polygibbs is the old gibbs1 fit mode, no longer used.

Default: 3.

- **set noefit**

Do not output input vs. fitted energies.

- **set noplotdh**

Do not write the static DeltaH plot.

- **set newpts**

If data is input in p(V) format, newpts is the number of volume points in the newly generated E(V) data. Default: 20.

- **set notrans**

Do not calculate phase transition pressures.

- **set errorbarerror_barerrorbarserrorBars**

Calculate error bars for dynamic properties, when polynomial averages are used.

- **set facexpand**

When data is input in p(V) format, it usually does not contain negative pressure ($V > V_0$) information, which is necessary for the operation of gibbs2. The original volume range in the input ($v_0 \rightarrow v_1$) is expanded to $v_0 \rightarrow v_2$ where

$$v_2 = v_1 + \text{facexpand} * (v_1 - v_0)$$

Default: facexpand = 0.40.

- **set phonfit {linear|spline}**

Type of interpolation of Phonon DOS with volume. Either linear or cubic not-a-knot spline. Linear interpolation is almost equivalent to the spline for a reasonably fine volume grid, and much faster.

Default: linear.

- **set writelevel {0|1|2}**

- 0: no output files except stdout.
- 1: only eos and eos_static.
- 2: all.

6 6 Test cases

Some test cases that can also serve as templates are provided together with gibbs2. They corresponds roughly to the examples in the section 'The structure of the input and output files', where they are described extensively. In addition, a directory dat/ contains some test datasets. Volume, energy and vibrational density of states have been provided for the volumes in the grid. In addition, the free energy contribution of the electrons has been calculated in Al. The corresponding summary files have extension 'res'.

- al_lda, al_pbe: fcc aluminium using the named xc functionals, 43 volume points.
- c_lda, c_pbe: diamond, 31 volume points.
- mgo_lda, mgo_pbe: MgO, phase B1, 174 volume points.
- mgob2_lda, mgob2_pbe: MgO, phase B2, 174 volume points.

The test cases are:

- 01_simple.ing: a simple example.
- 02_fits.ing: comparison of energy fit expressions.
- 03_tmodels.ing: different temperature models.
- 04_eec.ing: empirical corrections of the energy.
- 05_elec.gnu, 05_elec.ing: electronic contribution. The gnu file represents F_{el} and $-T \cdot S_{el}$ versus the input volume.
- 06_phases.ing: MgO phase transition.

To run them, simply do:

```
1 gibbs2 xx.ing xx.out
```

The xx.out is the main gibbs2 output file.

7 7 Citation of this program

Please, consider citing this program using references [fit1] and [impl] . If polynomial average fits are used, references [fit1] and [fit2] are relevant. Energy corrections are described in [corr] and energy corrections are polynomial fits are used in [rig] . The original gibbs article is [orig].

8 8 Copyright notice for gibbs2

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8.1 8.1 SLATEC copyright notice

The files in src/slatec are part of the SLATEC SLATEC Common Mathematical Library, Version 4.1, July 1993: a comprehensive software library containing over 1400 general purpose mathematical and statistical routines written in Fortran 77. See the src/slatec/ directory for more information.

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8.3 8.3 PPPack copyright notice

A library for the fitting of piecewise polynomials. Downloaded from netlib.org.

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