# QE-2019: Hands-on session - Day-2



#### Topics of Day-2 hands-on session:

- Structural optimizations:
  - relaxations atomic positions only (example1.relax)
  - variable-cell relaxations (example2.vc-relax)
- NEB method: saddle points of elementary chemical reactions (example3.neb/)
- Advanced functionals (example4.functionals/)
   (slides for advanced-functionals are provided by handson-day2-functionals.pdf)
- Automating the workflow with PWTK (example5.pwtk)

To get the latest version of the exercises, move to Day-2/ directory and execute:

```
$ git pull
```

## How to run calculations remotely on the "nsc" HPC cluster

Several utility commands have been implemented specially for the QE-2019 school to aid at submitting jobs to HPC cluster (login node: percolator.ijs.si). These are:

• remote\_mpirun — this is like mpirun, but it automatically submits the calculation to a queuing system on the "nsc" HPC system. For example, a pw.x calculation can be submitted as:

```
$ remote_mpirun pw.x -inp pw.file.in
```

where pw.file.in is the name of the pw.x input file. **BEWARE**: stdin/stdout redirection does not work for remote\_mpirun, hence you must use -inp option (i.e., do note use "<" redirection operator). You do not need to specify the number of processors, because the default is set to -np 8.

remote\_pwtk – this automatically submits the PWTK script to queuing system on the "nsc" HPC system. Example:

```
$ remote_pwtk script.pwtk
```

where script.pwtk is the name of the PWTK script.

• remote\_sbatch – automatically submits the Unix-shell script to queuing system on the "nsc" HPC system. Example:

```
$ remote_sbatch script.sh
```

where script.sh is the name of the Unix-shell script.

- nsc this makes ssh to "nsc" HPC login node (percolator.ijs.si), such that the user will be located in the same directory as used locally
- rsync\_to\_nsc copies specified files to the "nsc" cluster (percolator.ijs.si) to the same directory as is currently used locally. Example:

```
$ rsync_to_nsc *.in
```

This will copy all \*.in files from local directory to the same directory on the "nsc" cluster.

• rsync\_from\_nsc – download the specified file from the "nsc" cluster (percolator.ijs.si) from the same directory as is currently used locally. Example:

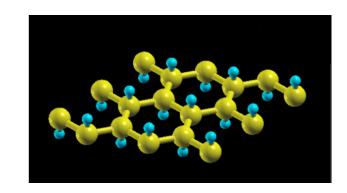
```
$ rsync_from_nsc *.out
```

This will copy all \*.out files from the "nsc" cluster.

## 1. How to perform structural optimization: graphane

• Move to Day-2/example1.relax/ directory.

Graphane is like graphene, with an H atom bound to each C atom in *trans* configuration. You need to optimize atomic positions, i.e., find the minimum-energy structure (zero forces).



- File pw.graphane.relax.in is a modified version of pw.graphene1x1.scf.in with:
  - calculation='relax' for structural optimization and a new namelist &IONS with variable upscale=100.0
  - ntyp=2 (2 types of atoms), nat=4 (4 atoms in the cell)
  - ATOMIC\_SPECIES card with 2 species of atoms and pseudopotentials
  - ATOMIC\_POSITION card with 4 initial positions (C–H distance  $\sim 1$  Å)

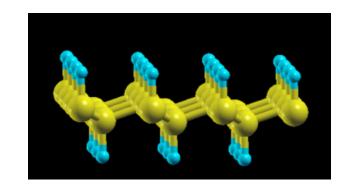
# 1. How to perform structural optimization: graphane (II)

• Run the structural optimization, i.e.:

```
$ pw.x < pw.graphane.relax.in > pw.graphane.relax.out &
```

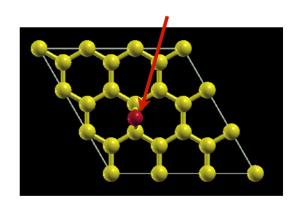
- When calculation finishes, analyze the output: it consists of several SCF steps, followed by calculation of forces and generation of new atomic positions.
- To visualize the evolution of the structure during structural optimization, execute:
  - \$ xcrysden --pwo pw.graphane.relax.out

Relaxed structure of graphane exhibits "buckling".



# 1. Supercell and structural optimization: graphene-oxide

• The first stage of graphene oxidation is the formation of an epoxy bridge. Let us add an O atom on a  $(3\times3)$  supercell of graphene



This example consists of two tasks: (i) build a  $(3 \times 3)$  supercell of graphene; (ii) add an O atom onto graphene– $(3 \times 3)$  supercell structure and run a relaxation calculation.

### Step-1:

- The  $(3 \times 3)$  supercell structure of graphene is provided in file pw.graphene3x3.scf.in, which is a modified version of pw.graphene1x1.scf.in. Please notice that:
  - Lattice parameters a and b are multiplied by 3: celldm(1)=13.962
  - Lattice parameter c remains the same, hence celldm(3), which equals c/a, is divided by 3, hence: celldm(3)=1.0.
  - There are 9 times the atoms of the original unit cell, i.e. nat=18.

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- Reciprocal lattice vectors in the xy plane are divided by 3 (look at the output), hence if you want the same k-point grid, just use K\_POINTS (automatic) with 3 3 1 0 0 0 grid, which is equivalent to 9 9 1 0 0 0 k-point grid of  $(1 \times 1)$  unit-cell
- Provided that the two k-point grids are equivalent:
  - \* The energy of the supercell  $E^{\rm SC} = 9E^{\rm UC}$  almost exactly (UC = unit cell)
  - \* all  $\epsilon^{\rm SC}({f k}_i)$  are (almost) equal to some  $\epsilon^{\rm UC}({f k}_j)$  if  ${f k}_j$  refolds into  ${f k}_i$

### Step-2:

- File pw.graphene3x3-0.relax.in is a modified version of pw.graphene3x3.scf.in with:
  - calculation='relax' for structural optimization and a new namelist &IONS
  - ntyp=2 (2 types of atoms), nat=19 (19 atoms in the cell)
  - ATOMIC\_SPECIES card with 2 species of atoms and pseudopotentials
  - ATOMIC\_POSITION card with 19 initial positions (C-O distance  $\sim 1.5$  Å)
- Run the structural optimization and analyze the output

# 2. How to perform variable-cell relaxation: hcp-Zinc

Zinc displays a hcp (hexagonal-closed-packed) crystal structure, hence it has two lattice parameters a and c. The unit-cell lattice vectors are:

$$\mathbf{a}_1 = (a, 0, 0), \quad \mathbf{a}_2 = (-\frac{a}{2}, \frac{a\sqrt{3}}{2}, 0), \quad \mathbf{a}_3 = (0, 0, c)$$

This lattice can be described as:

- ibrav=4, A=a, C=c, both in A, not a.u., as in file pw.Zn.scf.in
- or ibrav=4, celldm(1)=a, celldm(3)=c/a, as in the file pw.Zn.vc-relax.in

For hexagonal lattices one needs to optimize two lattice parameters. This can be done either manually or by using the variable-cell relaxation. (See also README.md).

1. **Manual way:** to optimize the a and c lattice parameters, one need to perform a 2D scan over the two parameters. With PWTK this can be achieved with the following snippet (full script is available in Zn-scan.pwtk):

```
foreach A [seq 2.4 0.1 2.8] {
   foreach C [seq 4.8 0.2 5.6] {
     SYSTEM " A = $A , C = $C"
     runPW pw.Zn.scf.$A.$C.in
   }
}
```

- 2. **Variable-cell relaxation:** This is a more convenient option. An example of how to perform variable-cell relaxation is provided by the input file pw.Zn.vc-relax.in. Notice:
  - calculation = 'vc-relax'
  - &IONS and &CELL namelists after the &ELECTRONS

To run the calculation, execute:

```
$ pw.x -in pw.Zn.vc-relax.in > pw.Zn.vc-relax.out
```

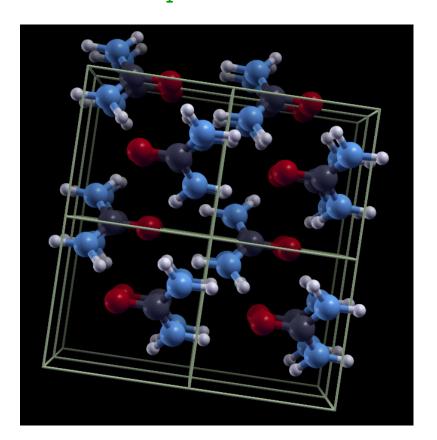
Inspect the output (file: pw.Zn.vc-relax.out) and notice that:

- several scf steps are performed, forces (zero by symmetry) and stresses computed
- the energy and the stress decrease as the minimum is approached
- a final scf step is performed with plane waves computed for the final cell
- the final cell is printed after the last CELL\_PARAMETERS card

Compare optimized parameters estimated from the *manual* 2D-scan to those obtained from the *variable-cell relaxation*.

# 2. Variable-cell relaxation (II): molecular crystal of Urea

While in the preceding example, the forces were zero by symmetry, in this example (pw.urea.vc-relax.in) both unit-cell and atomic positions are optimized by utilizing the *variable-cell relaxation* (calculation = 'vc-relax'). Beware that it is computationally heavier than the pw.Zn.vc-relax.in example.



See the instructions in README.md for how to run this calculation remotely.

# 3. NEB method: saddle points of elementary chemical reactions

Saddle points on the *Potential Energy Surface (PES)*, which correspond to *Transition States (TS)* of chemical reactions, can be found by means of the *Nudged Elastic Band (NEB)* method.

In the first example we will analyze a simple H transfer reaction:

$$H_2 + H \rightarrow H + H_2$$

### **Example 1a: No intermediate image**

In the NEB method one needs to supply a minimum of two images, the so-called **first**-and **last-image**, corresponding to reactants and products.

In neb.H2+H.in, you will see that near the end of the file FIRST\_IMAGE and LAST\_IMAGE are specified. Seven images are requested (num\_of\_images = 7) along the reaction and the neb.x code discretizes the path by means of a linear interpolation.

You can visualize this path with xcrysden:

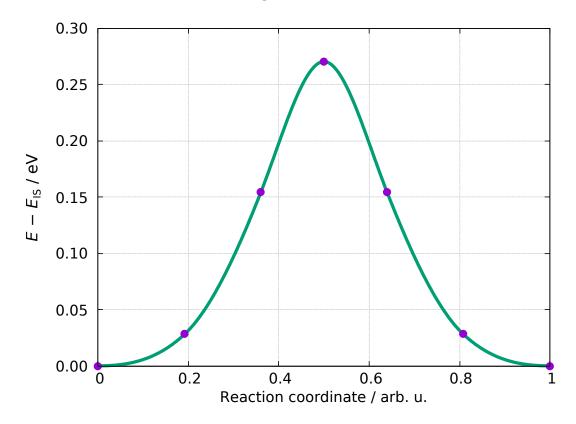
\$ xcrysden --pwi neb.H2+H.in

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To run the example execute:

When the calculation finishes analyze the output (neb.H2+H.out). Pay attention to the *number of steps* required to reach convergence and the reaction energy barrier.

The resulting plot should look something like this.



You can visualize individual points on the reaction path by using xcrysden.

Either by typing:

```
$ xcrysden --xyz H2+H.xyz
```

or

\$ xcrysden --axsf H2+H.axsf

#### **Example 1b: NEB with intermediate image**

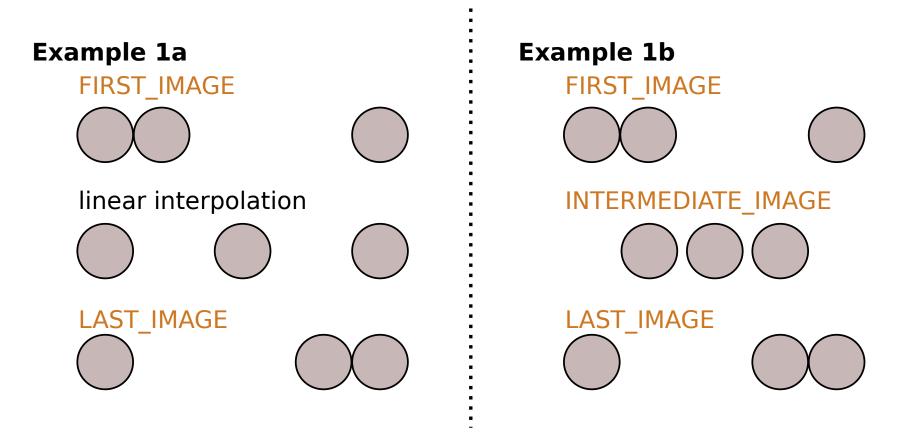
This example is similar as the previous example, however now chemical intuition is used to construct a better initial reaction path. This is achieved by using the INTERMEDIATE\_IMAGE card, in which a rough guess for the transition state is provided. You can visualize the path by using xcrysden:

```
$ xcrysden --pwi neb.H2+H.w-inter-image.in
```

Try to spot the difference between the two examples. Notice that without the intermediate image, the three atoms are evenly spaced in the middle of the initial interpolated reaction path, which is not a good guess of the transition-state. If we use INTERMEDIATE\_IMAGE then the three H atoms can be placed closer together, which

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is a lot closer to the actual transition state and the NEB calculation does not require as many steps to converge. This is shown schematically below:



To run this example type:

```
$ neb.x -inp neb.H2+H.w-inter-image.in > neb.H2+H.w-inter-image.out &
```

The final reaction barrier should be the same as in the previous calculation, however convergence should require fewer steps.

### **Example 2: A more complex NEB calculation**

In the second example we will study  $H_2$  dissociation on the Al(100) surface. We will be using INTERMEDIATE\_IMAGE and the calculation itself is also divided into two parts, in order to achieve convergence more quickly.

We will be using climbing-image NEB (CI-NEB), however initially we do not know which image is the *climbing-image* (the one with the highest energy), because its ID can change from iteration to iteration, depending on how far from convergence we are.

Thus we will first perform a plain NEB (CI\_scheme = 'no-CI') in order to stabilize the pathway and only once it is stabilized we will switch on the Cl-NEB with CI\_scheme = 'auto'. We will use a PWTK script to manage these two calculations.

#### **Steps:**

- Read neb.H2-diss.Al100-2x1-2L.in and try to understand it. This file will serve as the main input for the pwtk script. Visualize the initial guess by :
  - \$ xcrysden --pwi neb.H2-diss.Al100-2x1-2L.in
- Read the neb.pwtk script and try to understand it.

- Beware that this example takes about 20–30 min on a laptop. Hence, run the calculation remotely, i.e., submit it to the HPC cluster, i.e.:
  - \$ remote\_pwtk neb.pwtk
- To download the calculated output files, use:
  - \$ rsync\_from\_nsc \*.out

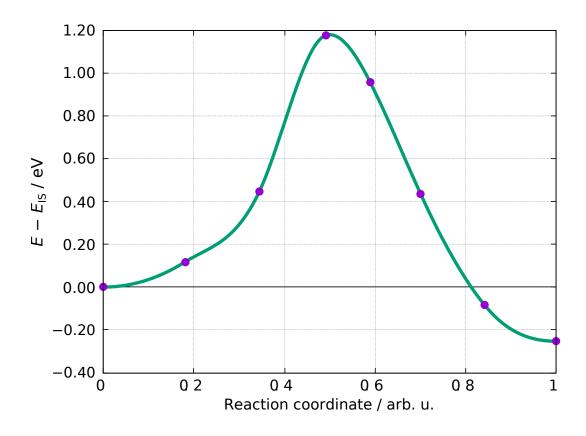
Give the remote computer some time to make the calculation. To download other data files that were produced by neb.x, use:

- \$ rsync\_from\_nsc .
- Once the calculation converges analyze neb.noCI.out and neb.auto.out and check the number of steps and the reaction energy barrier. The reaction energy graph can be plotted by:
  - \$ gnuplot H2-diss.Al100-2x1-2L.gp

and the corresponding path visulized by:

\$ xcrysden --axsf H2-diss.Al100-2x1-2L.axsf

The resulting PES along the reaction coordinate should look like this.



### 4. Advanced functionals

Slides for this exercise are provided by file handson-day2-functionals.pdf.

#### 5. PWTK: a short tutorial



PWTK is a Tcl-based scripting interface for Quantum ESPRESSO. It aims at providing a flexible and productive framework.

- PWTK web-site is:
  - http://pwtk.quantum-espresso.org/ or http://pwtk.ijs.si/
- PWTK documentation is available at:

```
http://pwtk.ijs.si/toc_index.html
(see also http://pwtk.ijs.si/pwtk-slides.pdf)
```

- Move to Day-2/example5.pwtk directory. Therein are three examples:
  - ex1.eos/ how to use EOS (equation-of-state) utility of PWTK
  - ex2.0@Al111/ how to run many calculations with a simple PWTK script
  - ex3.C0@Rh100/ a more elaborate PWTK example that shows how to glue together various calculations. In particular, it analyzes the bonding of CO molecule on Rh(100).

## 5. PWTK: hierarchical configuration

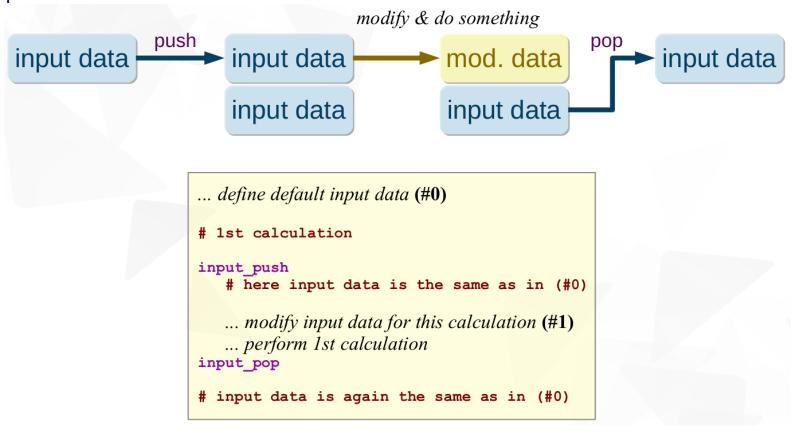
- Main user configuration file: \$HOME/.pwtk/pwtk.tcl
  - executables and how to run them (bin\_dir, prefix, postfix)
  - special directories (pseudo\_dir, outdir, wfcdir)
- Project-based configuration (via import), e.g,:
  - Day-2/example5.pwtk/common.pwtk
     Specification of default input data that are common to all examples in example5.pwtk/, e.g., cutoff energies, list of pseudopotentials, etc.
    - \* **Beware:** for each specific calculation PWTK filters atomic species and uses only those that are actually used. Due to that the index of species can change from case to case, hence for *ntyp*-type variables do not use numeric indices, but use atomic labels instead, e.g.:

```
starting_magnetization(1) = 1.0 (not recommended) starting_magnetization(Fe) = 1.0 (recommended)
```

- in a given example the project-based common.pwtk is then imported via, e.g., import ../common.pwtk

## 5. PWTK: concept of input data stacking

Input data stacking is a useful concept when performing multiple calculations. In such a case, one typically defines a set of default parameters for all calculations. However, each individual calculation may require some modification of the input data, yet it is inconvenient if such a change would affect the input data for other subsequent calculations. This inconvenience can be prevented with input\_push and input\_pop commands (actually it is more convenient to use input\_pushpop {...} instead). The concept is illustrated below:



## 5.1 PWTK: EOS utility

Move to Day-2/example5.pwtk/ex1.eos directory.

In this example the *equation-of-state* (EOS) will be calculated using pwtk. Run this calculation by typing:

```
$ pwtk eos.Rh-bulk.pwtk > eos.Rh-bulk.log &
```

This procedure performs 15 SCF calculations with different lattice parameters (using the initial guess as input) and then uses the results as input for ev.x, to calculate EOS in accordance with the four equations of state. The results are then summarized in eos.Rh-bulk.RESULTS (the \*.dat data-files with collected total energies are contained in eos.Rh-bulk.d/\*.dat).

OPTIMISED LATTICE PARAMETER, a0 (in Bohr units)::

		+	<b></b>	+	
# points		MURNAGHAN	BIRCH 0(1)	BIRCH 0(2)	KEANE
15: 14: 13: 12: 11: 10: 9:	115 114 214 213 313 312 412 411 511	7.294 7.294 7.293 7.294 7.293 7.293 7.293 7.293 7.293	7.293 7.293 7.293 7.293 7.293 7.293 7.293 7.293 7.293 7.293	7.293 7.293 7.293 7.293 7.293 7.293 7.293 7.293 7.293	7.294 7.294 7.294 7.294 7.293 7.293 7.297 7.296 7.296
		+	+	+	

Notice that the bulk-modulus is more sensitive to the sampling of data points, can you tell why?

OPTIMISED Total Energy, E0 (in Ryd units)::

# points	++   MURNAGHAN	BIRCH O(1)	•	KEANE
15: 115 14: 114 13: 214 12: 213 11: 313 10: 312 9: 412 8: 411 7: 511	-52.0363   -52.0363   -52.0364   -52.0364   -52.0364   -52.0364   -52.0364   -52.0364	-52.0364 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364	-52.0364 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364	-52.0363 -52.0363 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364 -52.0364
	++		+	+

OPTIMISED Bulk Modules, B0 (in kbar units)::

# points		MURNAGHAN	+	BIRCH 0(2)	KEANE
15: 14: 13: 12: 11: 10: 9: 8: 7:	115 114 214 213 313 312 412 411 511	2550 2557 2557 2568 2569 2587 2587 2587 2602 2603	2570   2571   2572   2578   2581   2594   2594   2606	2591 2597 2609 2616 2604 2593 2606 2602 2620	1097 1178 1207 1176 1244 1170 829 703 638
			+		

# 5.2 PWTK: gluing together many calculations

The subject of this example is the calculation of a 2D potential energy scan of lateral positions of O @ Al(111). It introduces the PWTK's concept of input-data stacking (i.e.  $input_pushpop {...}$ ).

The corresponding PWTK snippet is:

```
foreach ia [seq 0.0 0.2 1.0] {
   foreach ib [seq 0.0 0.2 1.0] {
     set x [expr $ia - 0.5*$ib]
     set y [expr sqrt(3)/2*$ib]

     input_pushpop {
        insertAtoms begin "0 $x $y 1.35 0 0 1"
        runPW pw.0-Al111.$ia.$ib.in
     }
   }
}
```

Beware that without the use of input-data stacking, each new iteration would add a new O atom to the structure. But with  $input_pushpop \{...\}$ , we add an O atom, make a calculation, and pop the O atom away.

The full script is provided by scan.pwtk; read it and and try to understand it.

**Beware:** this example takes quite long! Hence, it is better to submit it to HPC cluster. This can be achieved by:

\$ remote\_pwtk scan.pwtk

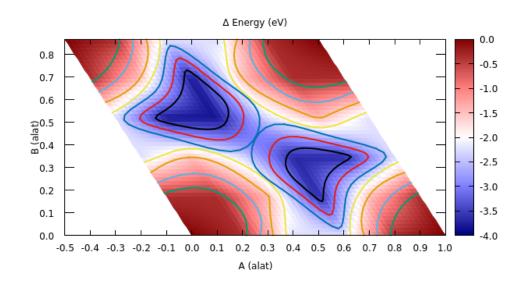
After calculation finishes, download the output and data files as:

\$ rsync\_from\_nsc .

The resulting potential-energy-surface can be visualized as:

\$ gnuplot plot2D.gp

Then plot the result with gnuplot. You should obtain something like this.



# 5.3 PWTK: analysis of electronic structure of CO@Rh(100)

This is a bit more elaborate PWTK example that shows how to glue together various calculations. In particular, it analyzes the bonding of CO molecule on Rh(100) by means of (i) charge-density difference, (ii) PDOS to atomic orbitals, (iii) MOPDOS to molecular orbitals, and (iv) ILDOS (integrated-local density of states).

The analysis reveals the charge-donation from the CO  $\sigma$  HOMO orbital into metal states and back-donation of charge from metal states into the CO  $\pi^*$  LUMO orbital.

The master PWTK script that performs the whole analysis is run-all.pwtk. This file imports several script, each performing specific task(s), in particular:

- relax.pwtk script for relaxing the CO@Rh(100) structure
- difden.pwtk script for calculating charge-density difference
- ildos.pwtk script for calculation of ILDOSes
- pdos.pwtk script for calculating PDOS to atomic orbitals, MOPDOS to molecular-orbitals of CO, and plots of molecular-orbitals ( $\psi^2$ ) of CO (BEWARE: this is a more elaborate script)

## 5.3.1 DIFDEN utility

PWTK has a *DIFDEN* utility to aid at calculating, e.g., charge-density difference:

$$\Delta \rho_{AB}(\mathbf{r}) = \rho_{AB}(\mathbf{r}) - \rho_{A}(\mathbf{r}) - \rho_{B}(\mathbf{r})$$

(remark: with DIFDEN you can calculate any difference, does not need to be density)

The above difference can be calculated using the following PWTK snippet (the CO@Rh(100) example):

```
DIFDEN {
    segment(1) = "all"
    weight(1) = 1.0
    name(1) = "all"

    segment(2) = "1 2"
    weight(2) = -1.0
    name(2) = "CO"

    segment(3) = "3-"
    weight(3) = -1.0
    name(3) = "Rh100"
}
difden_run difden.CO-Rh(100)
```

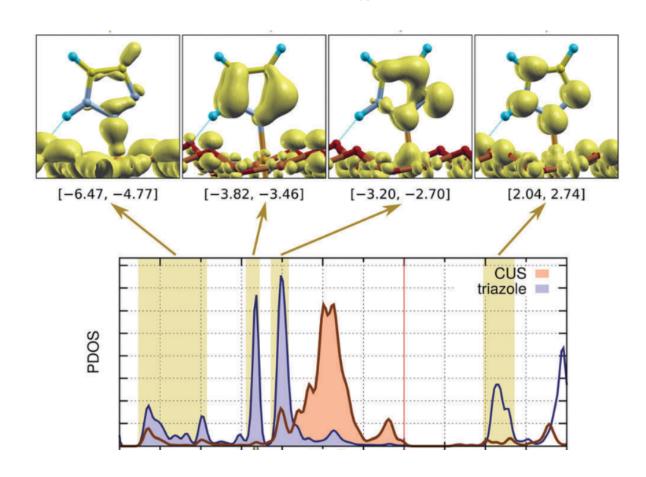
Note that the calculation of  $\Delta \rho(r)$  requires 6 or 7 calculations, i.e., pw.x and pp.x calculations for each of AB, A, and B. The last  $(7^{\rm th})$  pp.x calculation is to subtract the calculated densities. PWTK manages all these 7 calculations automatically.

The full PWTK script for calculating the  $\Delta \rho(r)$  of CO@Rh(100) is provided by file difden.pwtk.

### 5.3.2 ILDOS

ILDOS = integrated local density of states:

$$N(E_{\min}, E_{\max}, \mathbf{r}) = \int_{E_{\min}}^{E_{\max}} n(\epsilon, \mathbf{r}) d\epsilon = \sum_{v} \sum_{\mathbf{k}} \int_{E_{\min}}^{E_{\max}} |\psi_{v, \mathbf{k}}(\mathbf{r})|^2 \times \delta(\epsilon - \epsilon_{v, \mathbf{k}}) d\epsilon.$$



### 5.3.2 ILDOS

With aid of PWTK, a series of ILDOSes can be calculated using the following snippet:

```
INPUTPP " plot_num = 10 "

foreach {Emin Emax} {
    -6.00 -5.50
    -3.40 -3.30
    -3.05 -2.95
    ...
} {
    INPUTPP " emin = $Emin, emax = $Emax "
    PLOT " fileout = 'ildos.${Emin}.${Emax}.xsf' "
    runPP pp.ildos.${Emin}.${Emax}.in
}
```

The full PWTK script for calculating a series of ILDOSes of CO@Rh(100) is provided by file ildos.pwtk.

### 5.3.3 PDOS and MOPDOS

The PWTK script pdos.pwtk is a bit more lengthy/elaborate. It calculates PDOS (projected-density-of-states) to atomic orbitals, MOPDOS (molecular-orbital projected-density-of-states) to molecular-orbitals of CO, and it plots molecular-orbitals  $(\operatorname{sign}(\psi(\boldsymbol{r}))\psi^2(\boldsymbol{r}))$  of CO.

For further details, see README.md and the pdos.pwtk script itself. The purpose of this script is to demonstrate a bit more advanced usage of PWTK to those who may be interested.

Example: Day-2/example5.pwtk/ex3.C0@Rh100