# Coarse-graining for the Analysis of Soft Matter Scattering

submitted by

### Andrew R. McCluskey

for the degree of Doctor of Philosophy

of the

#### UNIVERSITY OF BATH

Department of Chemistry

August, 2018

#### **COPYRIGHT**

Attention is drawn to the fact that copyright of this thesis rests with the author. A copy of this thesis has been supplied on condition that anyone who consults it is understood to recognise that its copyright rests with the author and that they must not copy it or use material from it except as permitted by law or with the consent of the author.

This thesis may be made available for consultation within the University Library and may be photocopied or lent to other libraries for the purposes of consulation.





### **Declaration of Authorship**

I, Andrew R. McCluskey, declare that this thesis titled, "Coarse-graining for the Analysis of Soft Matter Scattering" and the work presented in it are my own. I confirm that:

- where the thesis or any part of the thesis such as a published paper, has been produced jointly with others, that a substantial part is the original work of myself, and
- where the thesis incorporates material already submitted for another degree, the extent of that material and the degree, if any, obtained.

Signed:			
Date:			

"Atticus told me to delete the adjectives and I'd have the facts."

Scout Finch – To Kill a Mockingbird

#### UNIVERSITY OF BATH

### Abstract

Department of Chemistry

Doctor of Philosophy

Coarse-graining for the Analysis of Soft Matter Scattering

by Andrew R. McCluskey

# Acknowledgements

## **Contents**

D	eclara	tion of	Authorship	iii
A	bstrac	ct		vii
A	cknov	wledge	ments	ix
1	The	ory		1
	1.1	Scatte	ring	1
		1.1.1	The scattering vector	
		1.1.2	Model-dependent analysis	
		1.1.3	Reflectometry	
		1.1.4	Small angle scattering	7
		1.1.5	Grazing incidence small angle scattering	7
	1.2	Probin	ng radiation	
		1.2.1	The generation of X-ray and neutrons	7
		1.2.2	Contrast variation	
	1.3	Optin	nisation & Sampling	13

### List of Abbreviations

MD molecular dynamics

C<sub>10</sub>TAB decyltrimethylammonium bromide dipalmitoylphosphatidylcholine

**SAS** small angle scattering

**GISAS** grazing-incidence small angle scattering

**SAXS** small angle X-ray scattering

**XRR** X-ray reflectivity

**GISAXS** grazing-incidence small angle X-ray scattering

**SANS** small angle neutron scattering

**NR** neutron reflectivity

GISANS grazing-incidence small angle neutron scattering

**DLS** Diamond Light Source

**ESRF** European Synchrotron Radiation Facility

Linac linear accelerator
BM bending magnet
rf radio-frequency cavity

**ID** insertion device

ILL Institut Laue-LangevinESS European Spallation Source

**EPSR** emperical potential structure refinement distorted wave Born approximation

PCFF poly consistent force field PBC periodic boundary condition

**OPLS** optimized potentials for liquid simulations

**NVE** constant number of particles, volume, and energy

NPT constant number of particles, pressure, and temperature NVT constant number of particles, volume, and temperature

WPEP whole particle effective potential C<sub>10</sub>TANO<sub>3</sub> decyltrimethylammonium nitrate

# **Physical Constants**

 $\pi = 3.14159\dots$ 

Speed of light  $c = 2.998 \times 10^8 \, \mathrm{m \, s^{-1}}$  Planck's constant  $h = 6.626 \times 10^{-34} \, \mathrm{J \, s}$ 

Golden ratio  $\Phi = 1.61803...$ 

Boltzmann's constant  $k_B = 1.380\,648 \times 10^{-23}\,\mathrm{J\,K^{-1}}$ 

# **List of Symbols**

$a_0$	optimum head-group area	$\mathrm{m}^2$
b	scattering length	m
b	bond length	m
$b_0$	equilibrium bond length	m
$c_{\alpha/\beta}$	atom concentrations	$\mathrm{m}^{-3}$
$d_n$	thickness of layer $n$	m
f	force	$ m kgms^{-2}$
$f_s$	scale factor	
$g_{lphaeta}$	partial pair distribution function	
g(X)	probability density function	
i	atom type	
$k_n$	wavevector for layer $n$	
$k_N$	dissociation constant from aggregate of $N$	$s^{-1}$
$l_0$	chain length	m
m	mass	kg
n	number of scattering vectors	
$n_i$	refractive index	
$q_i$	charge of atom $i$	$k_e$
p	surfactant packing parameter	
$r_c$	cut-off distance	m
$r_{ij}$	atomic distance	Å
$r_{n,n+1}$	Fresnel equation coefficient	
$r_{12}$	distance between surfactant centres-of-mass	Å
s	surfactant number	
t	timestep	$\mathbf{S}$
$t_F$	time-of-flight	S
u	potential energy	$kJ  mol^{-1}$
v	velocity	$\mathrm{m}\mathrm{s}^{-1}$
A	illuminated surface	$\mathrm{m}^2$
$A_{1,2,3}$	dihedral angle parameters	$kcal  mol^{-1}$
B	resultant matrix	
C	total solute concentration	$ m moldm^{-3}$
$C_s$	tail carbon atom in surfactant $s$	
D	number density of particles	$\mathrm{m}^{-3}$
$E_k$	kinetic energy	J
$E_{new}$	new energy	kJ
$E_{tot}$	total energy	kJ
$F(\mathbf{Q})$	diffuse scattering factors	
G(r)	radial distribution function	_
$I^(Q)$	scattering intensity	$\mathrm{cm}^{-1}$
K	equilibrium constant	_
$K_b$	bond force constant	$kcal  mol^{-1}  \mathring{A}^{-1}$
$K_{\theta}$	angle force constant	$kcal  mol^{-1}$

$L_F \ M$	distance of neutron flight layer matrix	m
N	aggregation number	molecule
$N_{at}$	number of atoms	
$N_{ m cycles}$	number of cycles	
$N_{\rm particles}$	number of particles	
$N_P$	number of undulator magnets	
$N_s$	head nitrogen atom in surfactant $s$	
P	probability	
P(Q)	particle form factor	1
Q	scattering vector magnitude	$\mathrm{m}^{-1}$
R	radius	m
$R_g$	radius of gyration	m
R(Q)	reflectivity resolution function	
Res(Q) $S$	nuclear spin quantum number	
$S_a$	surface area	$\mathrm{m}^2$
S(Q)	system structure factor	111
T	temperature	K
$T_{ m inst}$	instantaneous temperature	K
$T_{i,f}$	Fresnal transmission factor	
V	volume	$m^3$
$V_c$	chain volume	$m^3$
$V_p$	particle volume	meter <sup>3</sup>
$X_N$	concentration of molecules in aggregate of $N$	$mol dm^{-3}$
$\mathbf{k}_i$	incident wavevector	$m^{-1}$
$\mathbf{k}_f$	final wavevector	$\mathrm{m}^{-1}$
r	atomic position	$\mathrm{m}^{-1}$
Q	scattering vector	Ш
$\beta$	phase factor	
$eta_c$	fraction of c	
$\delta_{lphaeta}$	Kronecker $\delta$ -function	1 1 1-1
$rac{\epsilon_{ij}}{ heta}$	L-J well depth	kcal mol <sup>-1</sup>
$\theta$	polar angle angle	rad deg
$\theta_{1/2}$	surfactant- $r_{12}$ angle	deg
$ heta_c$	critical angle	rad
$\theta_e$	electron-photon angle	rad
$\theta_0$	equilibrium angle	deg
$\lambda$	wavelength	m
$\lambda_P$	magnetic period length	m
$\mu$	atomic mass	amu
$\mu_N$	mean chemical potential of aggregate of $N$	$J  \text{molecule}^{-1}$
$\mu_N^{\circ}$	mean interaction energy of aggregate of $N$	$J  \text{molecule}^{-1}$
$ u_{samp}$	sampling frequency	9
ho	scattering length density	$m^{-2}$
$\rho_0$	atomic density	$\mathrm{m}^{-3}$
$\sigma$	interfacial roughness	$^{\mathrm{m}}$ $\mathrm{cm}^{-1}$
$\sigma_i$	statistical uncertainty in $I^{\exp}(Q)$ distance of L-J minima	Å
$\sigma_{ij}$	distance of L-J Illillilla	Λ

$\sigma_{ m coh}$	coherent scattering cross section	$\mathrm{m}^2$
$\sigma_{ m incoh}$	incoherent scattering cross section	$\mathrm{m}^2$
$\phi$	azimuthal angle	$\operatorname{rad}$
$\phi$	dihedral angle	$\operatorname{rad}$
$\chi^2$	chi-squared	
$\omega$	neutron frequency	$s^{-1}$
$\omega_i$	incident frequency	Hz
$\omega_f$	final frequency	Hz
$\mathrm{d}\sigma/\mathrm{d}\omega$	differential cross-section	$\mathrm{m}^2$
$\Lambda$	temperature factor	

## 1 Theory

#### 1.1 Scattering

The use of scattering techniques to probe soft condensed matter systems is commonplace. In this work, we have focussed on the use of small angle scattering (SAS), reflectometry, and grazing incidence small angle scattering (GiSAS) techniques. These are particularly appropriate for application to soft condensed matter systems due to the length scales capable of being probed being similar to the persistence length of the soft condensed matter systems. The length scales covered for such techniques is from around 1 nm to 100 nm, as is shown in Figure 1.1. Since it is the equilibrium structures(s) under study, there is no interest in the system dynamics. Therefore, the system can be studied using exclusively elastic scattering techniques, where there is no energy transfer between the probing radiation and the system. This is in contrast to inelastic scattering where energy transfer occurs; facilitating the measurement of system dynamics, such as the dynamical modes of polymers of lipid bilayers.[1, 2] The techniques mentioned above all involve the use of elastic scattering and therefore probe the system equilibrium structure.

Both X-ray and neutron scattering techniques are discussed and used in this work. From an experimental viewpoint, there are significant differences between an X-ray scattering and a neutron scattering experiment. However, there is little variation in terms of the data analysis, where the differences are limited to; the nature of the scattering lengths, and the higher background that is present in the neutron scattering experiments.

#### 1.1.1 The scattering vector

The scattering of some probing radiation, by some sample can be represented as shown in Figure 1.2. Since only elastic scattering is being considered, there will be no change in the frequency of the radiation,  $\omega_i = \omega_f$ . This means that only the wavevector,  $\mathbf{k}$ , can change,  $\mathbf{k}_i \neq \mathbf{k}_f$ . The difference between the incident and final wavevectors is the scattering vector,  $\mathbf{q}$ , where,

$$\mathbf{q} = \mathbf{k}_i - \mathbf{k}_f. \tag{1.1}$$

The scattering vector strictly has units of  $m^{-1}$ , however it is often more practical to use  $nm^{-1}$  or  $\mathring{A}^{-1}$ . Throughout this work, units of reciprocal Ångstrom will be wherever possible. Since the frequency of the probing radiation does not change during an elastic scattering event, the wavelength,  $\lambda$ , will also not change, meaning that the moduli of the incident and final wavevectors are,

$$|\mathbf{k}_i| = |\mathbf{k}_f| = \frac{2\pi}{\lambda}.\tag{1.2}$$

This means that only the angle will change during the elastic scattering event. The vector diagram in Figure 1.3 can be used to describe the geometry of an elastic scattering event. From this, and Equation 1.2, the value of q, where  $q = |\mathbf{q}|$  can be shown

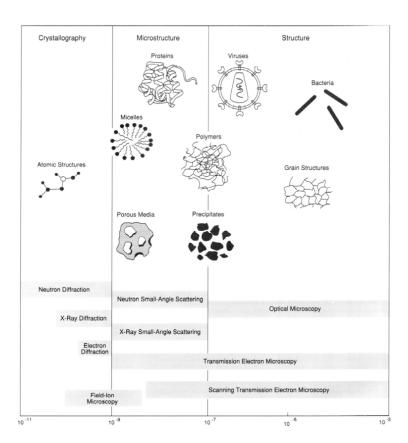


FIGURE 1.1: A representation of how different techniques can be used to probe various length scales, from Ref [3].

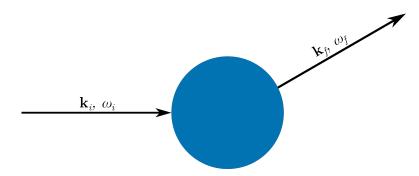


FIGURE 1.2: A schematic of the scattering of some probing radiation by a sample (blue circle), adapted from Ref [3].

1.1. Scattering 3

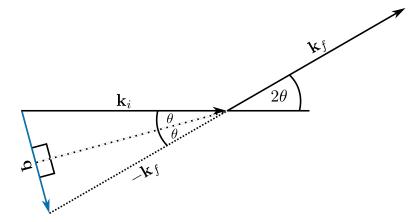


FIGURE 1.3: A vector diagram describing an elastic scattering event, adapted from Ref [3].

as, 
$$q = \frac{4\pi \sin \theta}{\lambda}.$$
 (1.3)

However, this fails to fully capture the three dimensional nature of the scattering event. Hence, it is necessary to describe the scattering with spherical coordinates,  $2\theta$ , and  $\phi$ , such that the incoming and outgoing radiation can be described as,

$$\mathbf{k}_{i} = \left(0, 0, \frac{2\pi}{\lambda}\right),$$

$$\mathbf{k}_{f} = \frac{2\pi}{\lambda} (\sin 2\theta \cos \phi, \sin 2\theta \sin \phi, \cos 2\theta),$$
(1.4)

where,  $|\mathbf{k}_f| = 2\pi/\lambda$ . This allows the scattering vector to be written,

$$\mathbf{q} = \frac{4\pi \sin \theta}{\lambda} (-\cos \theta \cos \phi, -\cos \theta \sin \phi, \sin \theta). \tag{1.5}$$

For an isotropic scattering pattern, it is the magnitude of the scattering vector, q, that is measured. In partical terms, the scattering vector allows for easy comparison of measurements made at different radiation wavelengths.

#### 1.1.2 Model-dependent analysis

All types of scattering patterns can be analysed by one of two methods; model independent and model-dependent. The nature of this work means that it will focus on model-dependent analysis methods, often where the model is derived from some atomistic, or coarse-grained simulation. Model-dependent analysis, has significant benefits over model-independent methods, such as improved resoution and more detailed information of the structure. However, the necessisty of the inclusion of *a priori* information within model-dependent analysis may act to bias the result, while this is undesirable, these assumptions can, and should, be educated based on the chemical information present – such as the propensity for twin-tailed lipid molecules to for monolayers at an air-water interface.[4]

The scattering from the model system is determined, using technique specific methods that are discuess in detail in laeter sections. This is then compared with the experimental data using some goodness-of-fit metric, as this the model is varied the quality of the fit, and therefore the value of the metric, will improve. In order to accurately

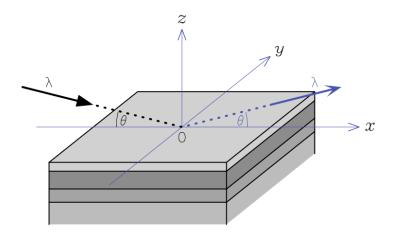


FIGURE 1.4: A schematic of specular reflectometry from a layered sample, from Ref [3].

reproduce the experimental measurement, it is necessary to include some instrumental resolution function, res(q), in the modelling procedure. This is instrument specific, although it may be approximated by convolving the experimental dataset with some Gaussian smearing function, the modelled intensity can then be determined from,

$$I(q) = res(q) * \frac{d\sigma(q)}{d\Omega}, \tag{1.6}$$

where,  $d\sigma(q)/d\Omega$  is the differential cross-section, a measure of the number of scattering particles hitting a given solid angle of the detector.

The aim of model-dependent analysis is to obtain a model for the system which agrees will with the experimentally measured scattering data, while producing something that is chemically, and physically relevant. This means that optimisation algorithm are often applied to these analysis problems, which are discussed, in detail, in Section 1.3.

#### 1.1.3 Reflectometry

Reflectometry involves the interaction of the probing radiation with some interface, from which the radiation is reflected. The geometry of a reflectometry experiment is shown in Figure 1.4, where the reflectometry instrument is in the horizontal configuration, ideal for the study of liquid interfaces. Reflectometry measurements give information about the structure perpendicular to the interface, the z-dimension in Figure 1.4, and therefore the analysis of reflectometry data is founding on the assumption that the layers will be completely homogenous in the plane of the interface, the xy-plane in Figure 1.4. In reality, since the layers are usually not completely homogeneous, an average is obtained for the area in the radiation beam. A reflectometry instrument operates by measuring the intensity of specular radiation at a series of different angles,  $\theta$ , or wavelengths,  $\lambda$ . The reflected intensity is defined in terms of q (by Eqn 1.3), and is defined as follows,

$$R(q) = \frac{\text{rate of specular reflective scattering}}{\text{rate of incidence}}.$$
 (1.7)

1.1. Scattering 5

It is clear from Equation 1.7 that the value of the measured reflectometry cannot be greater than one, as this would mean that more particles of probing radiation were being reflected than were incident.

#### **Analysis**

There are two model-dependent analysis techniques that can be applied to the rationalisation of a reflectometry dataset. The first is the kinematic approach,[3] where the reflectometry profile is modelled in terms of the scattering length density profile in the z-dimension,  $\rho(z)$ ,

$$R(q) \approx \frac{16\pi^2}{q^4} \left| \int_{-\infty}^{+\infty} \frac{\mathrm{d}\rho(z)}{\mathrm{d}z} \exp\left(-izq\right) \mathrm{d}z \right|^2, \tag{1.8}$$

where,  $d\rho(z)/dz$  is the first derivative of the scattering length density profile. However, this method has a significant problem, which can be described by applying Eqn 1.8 to the scattering length density profile of a bare silicon substrate, which can be modelled as a Heaviside function (Figure 1.5a),

$$\rho(z) = \begin{cases} 0, & \text{if } z < 0\\ \rho_{\text{Si}}, & \text{otherwise} \end{cases}$$
(1.9)

where,  $\rho_{Si}$  is the scattering length denisty of pure silicon (2.1 × 10<sup>-6</sup> Å<sup>-2</sup> for neutrons). The derivated of a stepwise Heaviside function is a scaled  $\delta$ -function (Figure 1.5b),

$$\rho'(z) = \rho_{Si}\delta(z). \tag{1.10}$$

Then, as in Eqn 1.8, the Fourier transform of this  $\delta$ -function is taken,

$$\rho_{\rm Si} \int_{-\infty}^{+\infty} \delta(z) \exp(-izq) dz = \rho_{\rm Si} \exp(0) = \rho_{\rm Si}. \tag{1.11}$$

This means that the reflectometry profile could be calculated from the following relationship,

$$R(q) \approx \frac{16\pi^2 \rho_{\rm Si}^2}{q^4}.$$
 (1.12)

The curve from this relationship is shown in Figure 1.5, where it is clear that the agreement with an experimental profile would be poor as  $q \to 0$ . It can be seen that for low values of q the calculated reflectometry is greater than 1, which voilates the physical constraint imposed with Eqn 1.7. This break down of the kinematic approach is due to the assumption present in this approach that the Born approximation will hold, that is that each particle of probing radiation will only be scattered once. However, in the reflectometry scattering geometry, this is no longer true rendering the kinematic approach invalid.

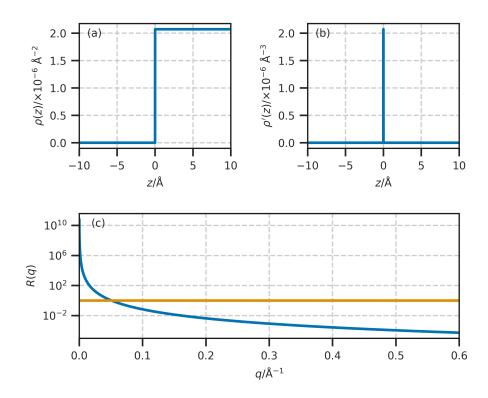


FIGURE 1.5: A graphical representation of the kinematic approach; (a) the Heaviside function describing the scattering length density profile of a bare silicon substrate, (b) the  $\delta$ -function arising from the first derivative of the function in (a), and (c) the reflectometry profile resulting from Eqn 1.8, where the orange line at R=1 identifies the break down between experimental and theory in the kinematic approach, adapted from Ref [3].

#### 1.1.4 Small angle scattering

#### 1.1.5 Grazing incidence small angle scattering

#### 1.2 Probing radiation

This work is focussed on the use of X-ray and neutron scattering, therefore it is pertinent to discuss how each of these probing radiation is produced and detail the advantages of each with resepect to the other.

#### 1.2.1 The generation of X-ray and neutrons

#### X-rays

X-rays are a form of electromagnetic radiation similar to visible light, albeit with a much shorter wavelength – from  $0.01\,\mathrm{nm}$  to  $10\,\mathrm{nm}$ . There are three common ways to produce X-rays; two are available within the laboratory, while the other is exclusive to large scale facilities.

The two laboratory source X-ray generation techniques are the X-ray tube and the rotating anode. An X-ray tube consists of a filament and an anode within a vacuum chamber, by passing a high voltage electrical current across the filament electrons are emitted which accelerate towards the anode. On collision with the anode, the rapid deceleration results in the emission of X-rays of a characteristic wavelength based on the anode material.[5] The most common material for an X-ray tube anode is copper which gives off radiation of about  $8 \, \mathrm{keV}$ .

The other common laboratory method for the generation of X-rays is the rotating anode, which is an improvment on the X-ray tube. In the X-ray tube, each time that an electron contacts the anode there is some energy transfer, this means that over many millions of collisions, the temperature of the anode can raise significantly – leading to a temperature limitation on the X-ray flux available. This lead to the development of the rotating anode, this is simply where the anode is made from a rotating wheel, so that the bombardment is spread across the whole wheel reducing the energy localisation. This allows an increase in the photon flux by about an order of magnitude.[5]

The third method of X-ray generation is at a synchrotron facility, this method has the drawback that it requires access to a national or international facility; such as Diamond Light Source (DLS) or the European Synchrotron Radiation Facility (ESRF). The way in which X-rays are generated at the synchrotron involves the acceleration of an electron, rather than the deceleration as with the laboratory sources. This is achieved by having relativistic electrons travel in around a curve, from Newtonian mechanics it is known that travelling on a curve at constant speed is equivalent to acceleration. This is achieved by firstly accelerating the electrons, produced in an linear accelerator (Linac), to near the speed of light in a booster synchrotron before injecting them into the storage ring. In the storage ring, the electrons are kept at relativistic speeds with bending magnets (BM) and straight sections making up a ring (Figure 1.6). The circularity of the ring is dependent on the number of bending magnets that make up the ring; for example, DLS has 48 bending magnets with 48 straight sections.

When an electron accelerates (or travels on a curve), Cherenkov radiation is emitted in accordance with the Cherenkov relation,

$$n_i \beta_c \cos \theta_e = 1, \tag{1.13}$$

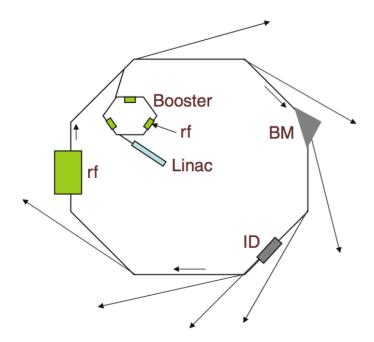


FIGURE 1.6: A schematic representation of a synchrotron radiation source, identifying the Linac, the booster ring, the radio-frequency cavities (rf), the bending magnet (BM) and the insertion device (ID), from Ref [6].

where,  $n_i$  is the refractive index for the dielectric medium,  $\beta_c$  is the fraction of the speed of light at which that electron is travelling, and  $\theta_e$  is the angle between the electron trajectory and the trajectory of the resulting photon.[6] The curve is the result of a bending magnet, meaning that at each bending magnet there can be a beamline which gives out synchrotron light. The light this is given off from a bending magnet is continuous and broad, covering a wide range of the electromagnetic spectrum. The alternative to a bending magnet beamline is a beamline which is served by an insertion device (ID). An insertion device is able to offer more specific radiation characteristics (photon energy, narrower band) than a bending magnet, and are placed on the magnet-free straight sections of the synchrotron. Common insertion devices include wavelength shifters, wigglers, and undulators.

The type of insertion device that is present at both I07 and I22 at DLS is an undulator. An undulator consists of a series of magnets of opposing polarity whihe causes the electrons to 'wiggle' back and forth (Figure 1.7). This results in a superposition of radition from  $N_P$  sources, where  $N_P$  si the number of magnets, yielding quasimonochromatic radiation. The brilliance of different X-ray sources are compared in Table 1.1, this shows the significant benefit that an undulator offers in terms of photon brilliance.

#### Neutrons

Neutrons hold an advantage over X-rays, particularly for application to the study of soft matter, in the ability to utilise contrast variation to increase the quantity of information from the sample, this is discussed in deatil in Section 1.2.2. However, neutrons cannot be produced safely on a laboratory scale, therefore it is always necessary to

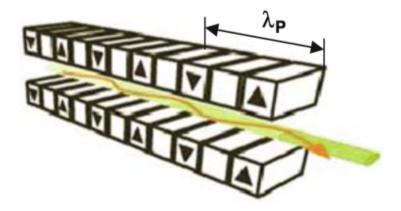


FIGURE 1.7: A diagram of an undulator insertion device such as that on I07 or I22 where  $\lambda_P$  is the period length between opposing magnets, from Ref [6].

TABLE 1.1: A comparision of the photon brilliance from different light sources, adapted from Ref [3].

Light source		
Candle	$10^{5}$	
X-ray tube	$10^{8}$	
Sun	$10^{10}$	
Bending magnet	$10^{15}$	
Undulator	$10^{20}$	

visit large scale facilities to harness neutrons for scattering experiments. These facilities come in two flavours; the reactor source and the spallation source, each offering unique benefits.

Neutron reactor sources, such as the Institut Laue-Langevin (ILL) in Grenoble, France, as currently the most common format of neutron source and are capable of producing the highest average neutron flux, the number of neutrons per second per unit area, for example the High-Flux Reactor at the ILL is capable of producing a neutron flux of  $1.5 \times 10^{15} \, {\rm neutrons \, s^{-1} cm^{-2}}$ .[7] A reactor source operates on the principle of nuclear fission, where an atomic nucleus is capable of breaking down into smaller nuclei, overcoming the strong nuclear force. This often involves using uranium enriched with its fissile isotope,  $^{235}$ U, which after the initial absorption of a stray neutron, from a cosmic ray, or spontaneous fission, will undergo fission to release, on average, 2.5 daughter neutrons, an example of a possible uranium fission mechanism is:

$$n + {}^{235}\mathrm{U} \longrightarrow {}^{236}\mathrm{U} \longrightarrow {}^{134}\mathrm{Xe} + {}^{100}\mathrm{Sr} + 2\,\mathrm{n}.$$

This type of mechanism is the basis for research, and nuclear power, reactors.[3] One of the major drawbacks for reactor neutron sources is the percieved public opinion towards such facilities. Major saftey concerns, such as "nuclear meltdown" and the resulting nuclear waste, mean that reactor souces are often unpopular and therefore struggle to obtain funding required for operation.

The other form of neutron source is a spallation source, this is much less controversial as it does not require fissile materials and hence there is no risk of a nuclear disaster. The ISIS neutron and muon source (Oxfordshire, UK) is an example of a spallation source, where high energy protons,  $800\,\mathrm{MeV}$ ,[8] are accelerated towards a tungsten target. When the protons strike the target, they can cause the release of a series of neutrons, the first batch of neutrons are given off with too high an energy to be useful, however, less excited neutrons are given off by secondary emissions. In addition to the public preception benefit, spallation sources also have a technological advantage in the time-of-flight technique. The time-of-flight (ToF) technique is based on the fact that at a spallation source, it is possible to know the time at which the neutron was ejected by the target to a high level of precision and accuracy, and therefore it is possible to measure the time taken for the neutron to reach the instrument. Since the neutron is a particle of a finite mass, m, it is possible to correlate the velocity, v, of the particle with the kinetic energy,  $E_k$ ,

$$E_k = \frac{mv^2}{2},\tag{1.14}$$

and with knowledge of the energy of the particle, its wavelength  $\lambda$ , can be determined by the de Broglie relation,

$$E = h\omega = \frac{hv}{\lambda},\tag{1.15}$$

where, h is Planck's constant and  $\omega$  is the neutron frequency. Therefore, the wavelength of the neutron is proportional to the inverse of the particle's velocity, and hece the time-of-flight,  $t_F$ ,

$$\lambda = \frac{h}{mv} = \frac{ht_F}{mL_F},\tag{1.16}$$

where,  $L_F$  is the distance between the target and the instrument. The fact that the neutrons can spread out in the flight from the target means that wavelength-dispersive techniques, where the neutron wavelength is measured rather than the scattering an-

gle, are possible at spallation sources which cannot be carried out at reactor sources. The negative side-effect of current spallation sources is that they have a lower average flux than reactor sources, however the building of the European Spallation Source (ESS) will change this as it offers an average flux similar to that of a reactor source, but with the benefits of the spallation technique.

A problem that is inherent for both reactor and spallation sources is that the energy of the neutrons given off is usually too high to be used to study condensed materials, such as soft matter. This means that moderation must be used to reduce the energy of the neutrons passing through the sample. The neutrons which are considered to be optimal for the study of condensed materials are thermal neutrons, named because their energy is appromately that of ambient temperature. Thermal neutrons are achieved by allowing the neutrons to pass through a large volume of moderator material, usually graphite or heavy water  $(D_2O)$ , stored at 300 K before they reach the instrument.[3]

#### 1.2.2 Contrast variation

The scattering profile generated by the interaction of some system with radiation depends on three factors:

- the spatial arrangement of the atoms in the system,
- the instrument being sued to measure the pattern instrumental resolution function, and
- the interaction between the radiation and the matter under investigation.

This final factor is perhaps better known as the 'scattering contrast', this is an extremely important factor in the study of soft matter, particularly when the probing radiation is the neutron. The scattering contrast makes it possible to select individual components of the system and investigate their structural properties.[9] The differential cross-section,  $d\sigma/d\Omega$  of a point scatterer varies only with respect to the scattering length of the species, b,

$$\frac{\mathrm{d}\sigma}{\mathrm{d}\Omega} = b^2. \tag{1.17}$$

However, when studying an ensemble of particles, it is easier to use the scattering length density,  $\rho$ ,

$$\rho = \frac{1}{V} \sum_{j} b_j \tag{1.18}$$

where,  $b_i$  is the coherent scattering length of all atoms in some volume, V.

When an X-ray interacts with an atom, it is scattered by the interaction with the electrons, this is due to the X-ray being a form of electromagnetic radiation. Furthermore, it means that the scattering length of an atom by an X-ray is directly proportional to the number of electrons in the atom, so it is therefore difficult to discern between the scattering from a carbon atom (6 electrons) and a nitrogen atom (7 electrons), furthermore the scattering from hydrogen atoms is practically non-existent.

The scattering length a neutron by an atom varies unsystematically with respect to the atomic number of a species, this is shown in Figure 1.8. Furthermore to the apparently random variation with changes in atomic number, there is also significant variation with mass number, e.g. between isotopes of the same atom. This is also dependance due to the magnetic state of the atom, however this is normally unimportant for soft matter. The scattering lengths differ with the nuclear spin energy level, this

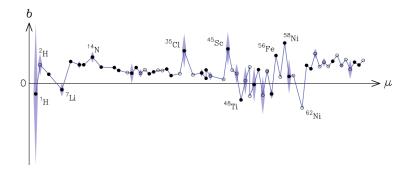


FIGURE 1.8: The variation of the average neutron scatteirng length,  $\langle b \rangle$  (circles), with atomic mass,  $\mu$ . The standard deviation,  $\Delta b$ , is indicated with the shaded regions, from Ref [3].

TABLE 1.2: Examples of coherent and incoherent scattering cross-sections, from Ref [9].

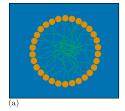
Isotope		$\sigma_{\rm coh}/10^{-28}{\rm m}$	$\sigma_{\rm incoh}/10^{-28}{\rm m}$
<sup>1</sup> H <sup>2</sup> H <sup>12</sup> C <sup>14</sup> N <sup>16</sup> O	1/2	1.8	79.7
$^{2}H$	1	5.6	2.0
<sup>12</sup> C	0	5.6	_
$^{14}N$	1	11.6	0.3
<sup>16</sup> O	0	4.2	_

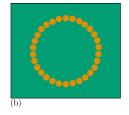
leads to an average scattering length,  $\langle b \rangle$ , for isotopes where the nuclear spin is non-zero ( $S \neq 0$ ). There are two forms of scattering, coherent and incoherent, for which the scattering cross-sections,  $\sigma$ , are determined by,

$$\sigma_{\text{coh}} = 4\pi \langle b \rangle^2$$

$$\sigma_{\text{incoh}} = 4\pi (\langle b^2 \rangle - \langle b \rangle^2)$$
(1.19)

The coherent scattering is the scattering from nuclei that all have the same value of  $\langle b \rangle$ , and leads to the important scattering pattern. Whereas, the incoherent scattering is caused by the 'disorder' between the isotopes, and is the cause of the background present in the measurement. Examples of these scattering cross-sections for nuclei relevant to soft matter are shown in Table 1.2. It can be seen that the incoherent scattering from the <sup>1</sup>H nuclei is more than forty times the coherent scattering. This leads to a large, intrusive background present in the scattering pattern of hydrogenous samples. The difference between the scattering of <sup>1</sup>H and <sup>2</sup>H, evident in Table 1.2, can lead to a very useful technique if soft matter scattering, known as contrast variation. The idea of contrast variation is based on the substitution of one isotope of an atom for another, while not introducing significant change to the properties of the material. Traditionally the benefit of this came in terms of contrast matching out a part of the system to reduce the dimensionality of the problem for analysis. For example, by matching the solvent scattering length density to that of the tails of the surfactants at the centre of a micelle there would only be scattering from the heads, and conversely there would only be scattering from the tails if the solvent had the same scattering length density as the head groups. This means that the problem becomes more straightforward as there are fewer variable parameters when fitting the data. This idea is represented graphically in Figure 1.9. The technique of contrast variation may also be used in terms of





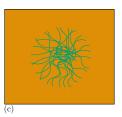


FIGURE 1.9: The effect of varying the scattering length density of the solvent in a micelle system, (a) the system in pure solvent, (b) the solvent is contrast matched to the surfactant tails, and (c) the solvent is contrast matched to the surfactant heads.

data analysis. By increasing the number of data sets corresponding to a single model – at different constrasts, the solution for the true sturcture of the model from the scattering data becomes more robust. This is due to the fact that each different contrast can be considered as an independent measurement of the same system, and hence each set of scattering data can be used within the data analysis procedure to obtain the best global agreement to the experiment. This co-refinement of multiple experiments can, under the right conditions, be used to simulatenously consider both neutron and X-ray datasets.[10]

There is also the possibility of using contrast variation when the probing radiation is the X-ray, through the use of anomalous scattering. This is where different wavelengths of radiation give different scattering, when the wavelengths are on opposite sides on an X-ray absorption edge. This is not frequenctly used for soft matter species, as the X-ray absorption edges for elements common in soft matter (H, C, N, O, etc.) are at very low X-ray energies so generally outside of the accessible range. [9]

#### 1.3 Optimisation & Sampling

# **List of Figures**

1.1	A representation of now different techniques can be used to probe var-	
	ious length scales, from Ref [3]	2
1.2	A schematic of the scattering of some probing radiation by a sample	
	(blue circle), adapted from Ref [3]	2
1.3	A vector diagram describing an elastic scattering event, adapted from	
	Ref [3]	3
1.4	A schematic of specular reflectometry from a layered sample, from Ref	
	[3]	4
1.5	A graphical representation of the kinematic approach; (a) the Heavi-	
	side function describing the scattering length density profile of a bare	
	silicon substrate, (b) the $\delta$ -function arising from the first derivative of	
	the function in (a), and (c) the reflectometry profile resulting from Eqn	
	1.8, where the orange line at $R = 1$ identifies the break down between	
	experimental and theory in the kinematic approach, adapted from Ref	
	[3]	6
1.6	A schematic representation of a synchrotron radiation source, identi-	
	fying the Linac, the booster ring, the radio-frequency cavities (rf), the	0
4 =	bending magnet (BM) and the insertion device (ID), from Ref [6]	8
1.7	A diagram of an undulator insertion device such as that on I07 or I22	0
1.0	where $\lambda_P$ is the period length between opposing magnets, from Ref [6].	9
1.8	The variation of the average neutron scatteirng length, $\langle b \rangle$ (circles), with	
	atomic mass, $\mu$ . The standard deviation, $\Delta b$ , is indicated with the shaded	10
1.0	regions, from Ref [3]	12
1.9	The effect of varying the scattering length density of the solvent in a	
	micelle system, (a) the system in pure solvent, (b) the solvent is contrast	
	matched to the surfactant tails, and (c) the solvent is contrast matched	10

## **List of Tables**

1.1	A comparision of the photon brilliance from different light sources, adapted	i
	from Ref [3]	9
1.2	Examples of coherent and incoherent scattering cross-sections, from Ref	
	[9]	12

# List of Algorithms

## **Bibliography**

- [1] V. García Sakai and A. Arbe, Curr. Opin. Colloid Interface Sci, 2009, 14, 381–390.
- [2] B. Farago, Curr. Opin. Colloid Interface Sci, 2009, 14, 391–395.
- [3] D. S. Sivia, *Elementary Scattering Theory*, Oxford University Press, Oxford, UK, 2011.
- [4] A. R. McCluskey and K. J. Edler, Curr. Org. Chem., 2018, 22, 750–757.
- [5] H. Schnablegger and Y. Singh, *The SAXS Guide: Getting acquainted with the principles*, Anton Paar GmbH., Graz, Austria, 4th edn., 2017.
- [6] M. C. Garcìa-Gutièrrez and D. R. Rueda, in *Applications of Synchrotron Light to Scattering and Diffraction in Materials and Life Sciences*, ed. T. A. Ezquerra, M. C. Garcìa-Gutièrrez, A. Nogales and M. Gomez, Springer-Verlag Berlin Heidelberg, Heidelberg, Germany, 1st edn., 2009, vol. 776, ch. 1, pp. 1–20.
- [7] ILL::Neutrons for Science::Technical characteristics, https://www.ill.eu/reactor-environment-safety/high-flux-reactor/technical-characteristics/, Accessed: 2016-08-08.
- [8] ISIS How ISIS works, http://www.isis.stfc.ac.uk/about/how-isis-works6313.html, Accessed: 2016-08-08.
- [9] P. Schurtenberger, in *Neutrons, X-rays and Light: Scattering Methods Applied to Soft Condensed Matter*, ed. P. Linder and T. Zemb, Elsevier, Amsterdam, The Netherlands, 1st edn., 2002, ch. 7, pp. 391–420.
- [10] A. Nelson, J. Appl. Crystallogr., 2006, 39, 273–276.