# **Nuclear Fuel Performance**

NE-533

Spring 2022

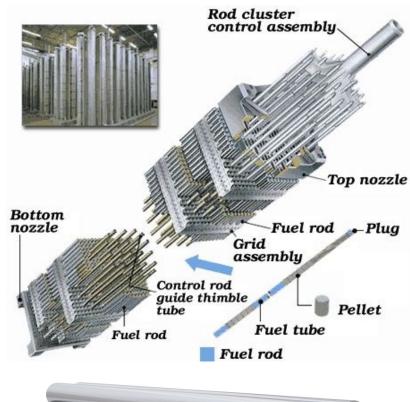
#### **Last Time**

- Hope everyone had a good spring break
- Graded exams will be sent out later today
- Questions on MOOSE project part 1?
- Discussed fission product families and effects
- Outlined different dimensional change effects

# ZIRCONIUM CLADDING

# Cladding

- The purpose of the cladding is to:
  - Hold the pellets together so that coolant can freely flow past
  - Transport heat from fuel to the coolant
  - Contain fission products
  - Contain fuel fragments
- We would also prefer if the cladding had little to no impact on the neutron transport in the reactor





# Why Zirconium alloys?

#### **Benefits**

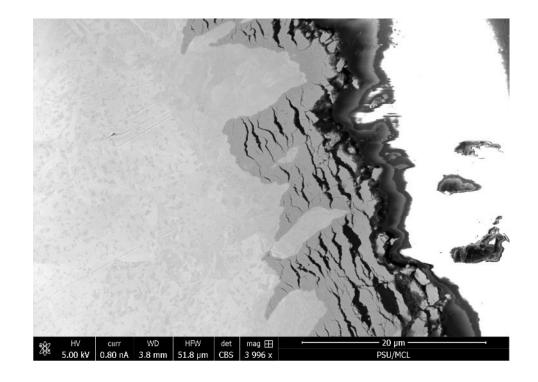
- Low neutron cross section
- Corrosion resistance in 300°C water
- Resistance to void swelling
- Adequate mechanical properties
- Good thermal conductivity
- Affordable cost
- Available in large quantities

#### **Problems?**

- Corrosion under high temperature steam
- Hydride embrittlement
- Anisotropic characteristics lead to creep and growth

### **Zirconium**

- Pure zirconium was not acceptable due to its oxidation behavior
- Oxygen with water reacted to form an oxide layer
- The oxide layer is brittle and flakes off, allowing more oxidation to occur
- Zr alloys were developed to reduce corrosion resistance

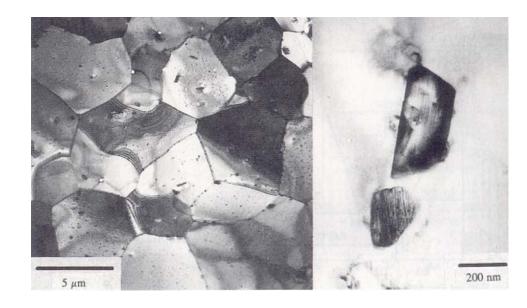


# **Commercial Zr Alloys in PWRs**

Alloy	Sn %	Nb %	Fe %	Cr %	Ni %	O %		
PWRs (structural components and fuel rods)								
Zircaloy-4 (SRA)	1.2-1.7	-	0.18-0.24	0.07-0.13	-	0.1-0.14		
ZIRLO (SRA)	1	1	0.1	-	-	0.12		
Optimized ZIRLO (pRXA)	0.7	1	0.1	-	-	0.12		
M5 (RXA)	-	0.8-1.2	0.015-0.06	-	-	0.09-0.12		
HPA-45 (SRA/RXA)	0.6	-	Fe+V	-	-	0.12		
NDA (SRA)	1	0.1	0.3	0.2		0.12		
MDA (SRA)	0.8	0.5	0.2	0.1		0.12		

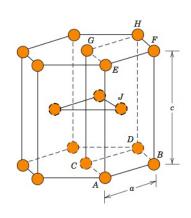
# **Alloying Elements**

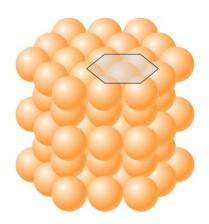
- The alloying elements impact the microstructure of the material
- The intermetallic precipitates significantly improve the oxidation behavior
- In Zircaloy 2, the precipitates are
  - Zr(Cr, Fe)<sub>2</sub>
  - Zr<sub>2</sub>(Ni, Fe)
- In Zircaloy 4, the precipitates are
  - Zr(Cr, Fe)<sub>2</sub>
- Phosphides (Zr<sub>3</sub>P) and silicides (Zr<sub>3</sub>Si) are also occasionally found
- The precipitate distribution, size, morphology, and composition all impact the properties of the material

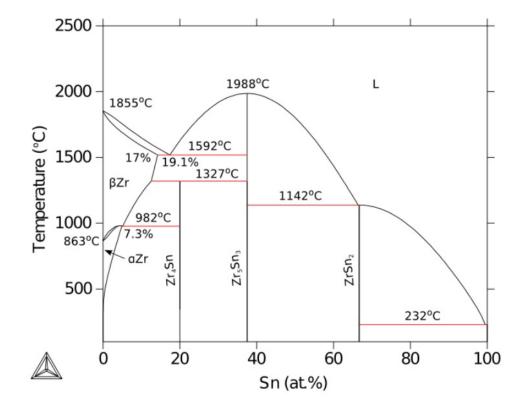


## **Zirconium Phases**

- The α-Zr phase has a hexagonal-closepacked (HCP) structure
  - At temperatures below about 863°C
  - Has the most desirable properties
- The β-Zr phase has a body-centered cubic (BCC) structure
  - We try to avoid this phase







### **Zr Tube Fabrication**

- The cladding tubes are fabricated using various processes that SEVERELY deform the material
- The severe plastic deformation causes significant dislocation hardening, that can make the tube material too brittle
- Plastic deformation along slip planes results in reorientation of the grains
- The properties of the material are controlled by heat treatment
- Raising the temperature, but below 863°C, anneals the sample to reduce cold work (stress-relieved)-SRA
- Raising the temperature above 863°C changes to the β phase. They then quench the sample to create a random texture in the α phase-RXA or pRXA

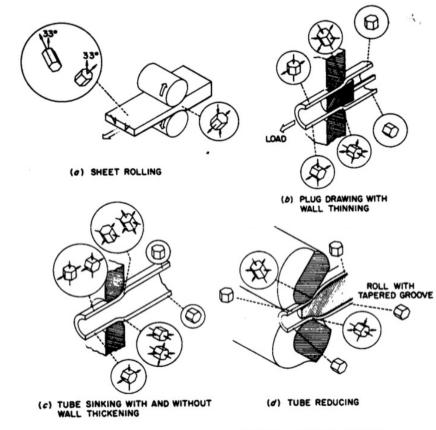
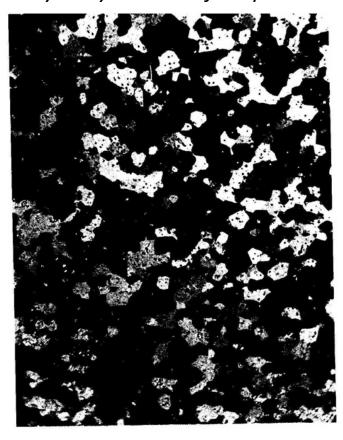


FIG. 48. Strain states for various types of fabrication of Zircaloy-2 [109].

# Zr alloy microstructures

Fully recrystallized after quench



Stress-relieved microstructure

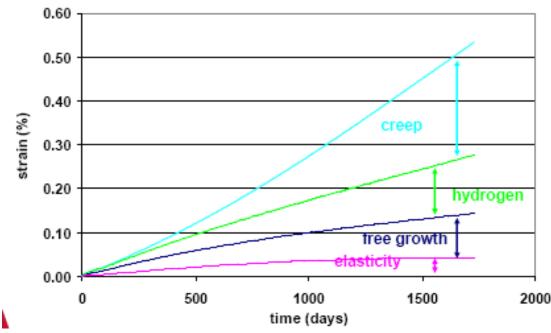


## **Zirconium Creep and Growth**

- The cladding undergoes a very large amount of irradiation during its life, the main effects are:
  - Interstitial loops form on prism planes
  - Later in the reactor exposure, vacancy loops form on the basal plane
- The overall number of displacements to the cladding can be calculated with the NRT model to be ~ 8 x 10<sup>-8</sup> dpa/s, which, over 3 years exposure gives a total of ~8 dpa (every atom in the solid is displaced on the average eight times)

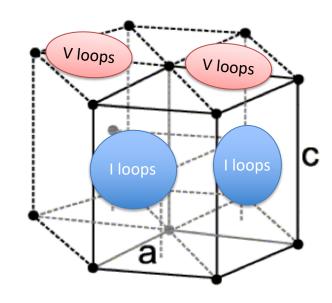
$$v(T) = \frac{\kappa E_D}{2E_d} = \frac{\kappa (T - \eta)}{2E_d},$$

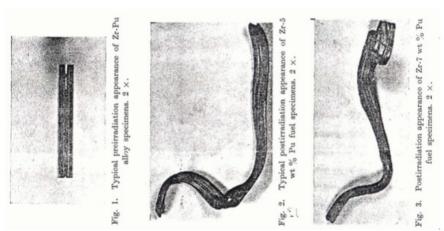
in-pile Zy4 grid at 325°C



### **Irradiation Growth**

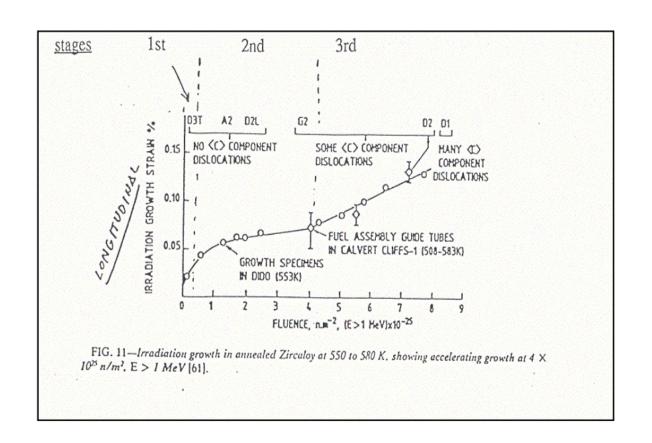
- Irradiation growth results from material anisotropy
- There must be anisotropy in the defect behavior within the unit cell
- There also must be a texture formed in the grain orientations
- Interstitial loops form on pyramidal planes, causing shrinkage along the center axis
- Later, vacancy loops form on basal planes, making it even worse
- The result is a change in shape, shrinking in the 0001 direction and expanding in the pyramidal direction





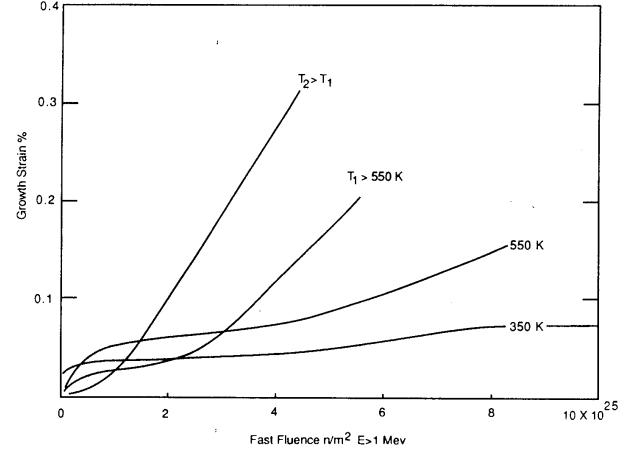
#### **Irradiation Growth**

- Three stages of irradiation growth
  - Initial rapid growth to small strains
  - Slow growth, gets skipped in cold worked material
  - Accelerated or breakaway growth, caused by formation of vacancy loops on the basal plane
- Irradiation growth is affected by:
  - Fluence, Cold work, Texture, Irradiation temperature, Material chemistry, Hydrogen effects



#### **Irradiation Growth**

- Higher temperature leads to earlier breakaway and higher growth
- Higher temperature adds mobility to defects, allowing them to diffuse faster, form clusters, and exacerbate the defect cluster induced anisotropic growth



## Creep

- Empirical models have been developed for thermal and irradiation creep of Zircaloy
- Both based on the Von Mises stress

$$\sigma_m = \sqrt{\frac{1}{2} \left( (\sigma_{11} - \sigma_{22})^2 + (\sigma_{22} - \sigma_{33})^2 + (\sigma_{22} - \sigma_{11})^2 + 6 \left( \sigma_{12}^2 + \sigma_{23}^2 + \sigma_{31}^2 \right) \right)}$$

- Thermal Creep  $\dot{\epsilon}_{ss} = A_0 \left(\frac{\sigma_m}{G}\right)^n e^{\left(\frac{-Q}{RT}\right)}$ 
  - With  $A_0 = 3.14 \times 10^{24}$  (1/s); shear modulus  $G = 4.2519 \times 10^{10} 2.2185 \times 10^7 T$  Pa; n = 5;  $Q = 2.7 \times 10^5$  J/mol
- Irradiation Creep  $\dot{\epsilon}_{ir} = C_0 \Phi^{C_1} \sigma_m^{C_2}$ 
  - φ is the fast neutron flux n/(cm<sup>2</sup> s) = 3E11xLHR n/(cm<sup>2</sup>-s)
  - Note that SRA stands for stress relief annealed
  - RXA for recrystallization annealed
  - PRXA stands for partially recrystallization annealed

Clad Type	$C_0$	$C_1$	$C_2$
SRA	$3.557 \times 10^{-24}$	0.85	1.0
RXA	$1.654 \times 10^{-24}$	0.85	1.0
PRXA	$2.714 \times 10^{-24}$	0.85	1.0
ZIRLO	$2.846 \times 10^{-24}$	0.85	1.0

## **Creep Example**

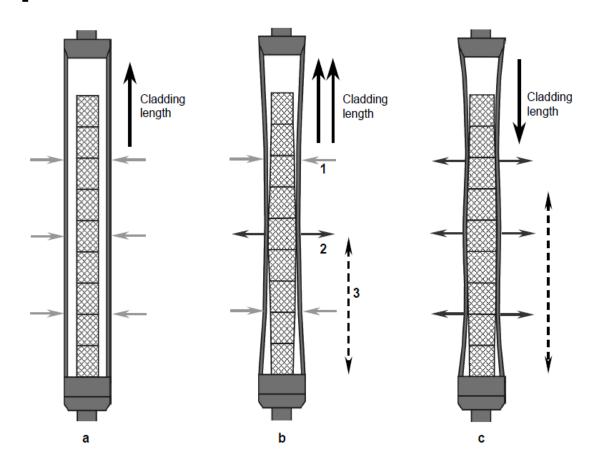
- Consider an SRA cladding tube at T = 600 K and LHR = 250 W/cm, with a stress  $\sigma_m$  = 200 MPa. What is the total creep strain after three years?
- First, we will calculate the thermal creep
  - $-A_0 = 3.14 \times 10^{24} (1/s)$
  - G =  $4.2519 \times 10^{10}$   $2.2185 \times 10^{7}$  Pa = 4.2519e10 2.2185e7\*600 = <math>2.92e10 Pa
  - $Q = 2.7 \times 10^5 \text{ J/mol}, n = 5, R = 8.3144598 \text{ J/(K mol)}$
  - $-3.14e24*(200/2.92e4)^5*exp(-2.7e5/(8.3144598*600)) = 1.48e-10$  1/s
- Now we will calculate the irradiation creep
  - $C_0 = 3.557e-24, C_1 = 0.85, C_2 = 1.0$
  - $-\Phi \approx 3e11 LHR = 3e11*250 = 7.5e13 n/(cm<sup>2</sup> s)$
  - $-3.557e-24*(7.5e13)^0.85*200^1 = 4.43e-10 1/s$
- The total creep strain rate is 1.48e-10 + 4.43e-10 = 5.91e-10 1/s
- The total creep strain after three years is (assuming constant conditions) 5.91e-10\*(3600\*24\*365\*3) = 0.056 = 5.6% strain

$$\dot{\epsilon}_{ss} = A_0 \left(\frac{\sigma_m}{G}\right)^n e^{\left(\frac{-Q}{RT}\right)}$$

$$\dot{\epsilon}_{ir} = C_0 \Phi^{C_1} \sigma_m^{C_2}$$

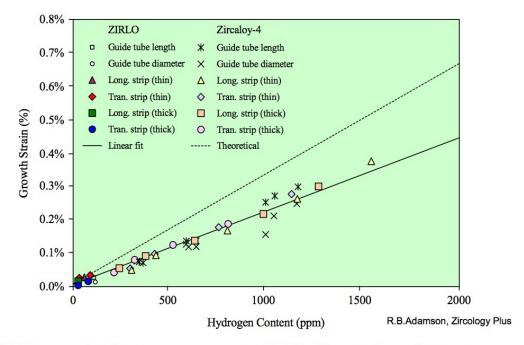
## Creep

- Creep impacts fuel performance by shrinking the gap and then conforming to the pellets
- a) before "fuel-cladding" interaction;
  - b) start of "fuel-cladding" interaction;
  - c) "fuel-cladding interaction" over most of fuel column
- (1)creep down from water pressure;
   (2)creep out from fuel column;
   (3)fuel column axial stress



# **Zirconium Hydrides**

- Hydrides cause size change because the hydride lattice is larger than the zirconium lattice
- There are a number of different Zr-H phases and morphologies that can present
- 0.1% H can cause 0.2% growth



1000 ppm hydrogen can cause 0.2% dimension change

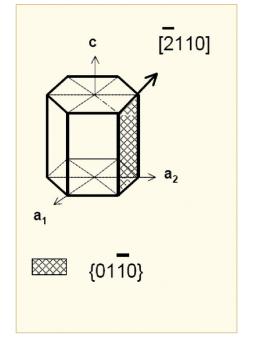
## **Summary of Creep and Irradiation Growth**

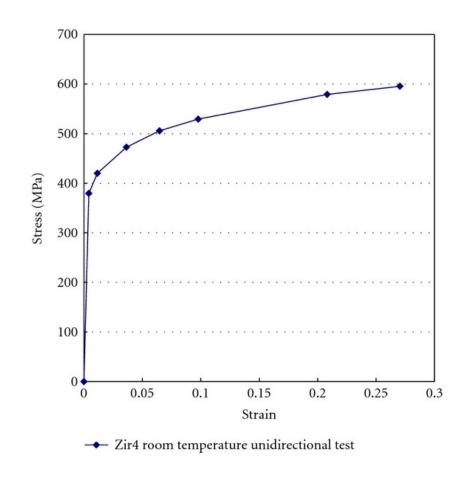
- Growth and creep are the major mechanisms for dimensional instability in zirconium alloy cladding
- Growth results from the clustering of interstitials on prismatic planes, and eventually from the clustering of vacancies on basal planes such that the material shrinks in the axial direction
- Growth depends on the fluence, coldwork, texture, temperature, and composition

## Mechanical Behavior of Zr Cladding

 Zirconium alloys plastically deform primarily due to dislocation motion on prismatic

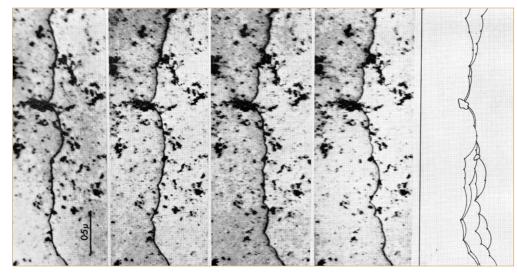
planes

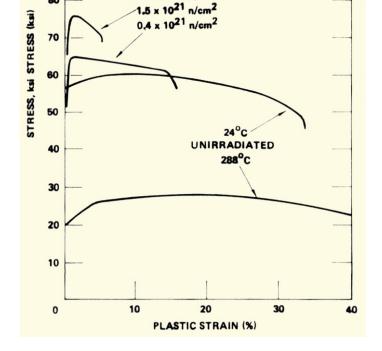




## **Irradiation Hardening**

- Under irradiation, the interstitial loops that buildup on the prismatic planes cause hardening
- The UTS goes up with fluence, while the strain before fracture goes down



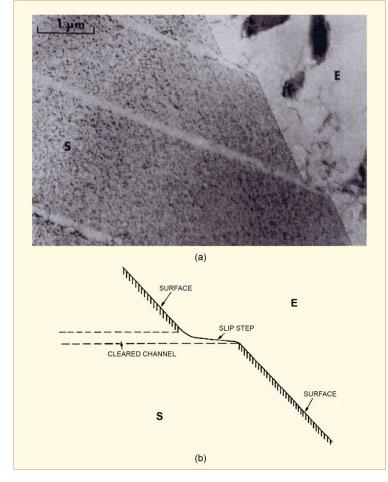


0.8 x 10<sup>21</sup> n/cm<sup>2</sup>

IRRADIATED, 288°C

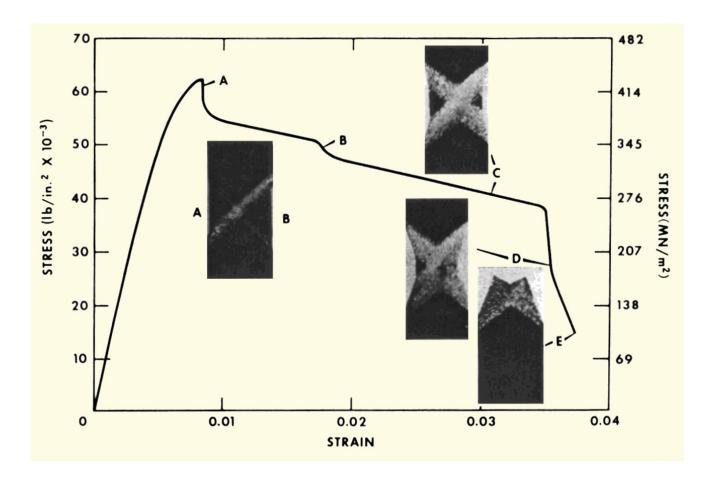
#### **Dislocation Channels**

- When stress is high enough, loops are "swept away", forming a clean channel
- Once a channel is cleaned out, dislocations continue to move along it
- This results in lots of deformation in that one area, creating slip steps
- In HCP zircaloy, channels form on basal planes or prism planes depending on
  - Load direction
  - Temperature
  - Oxygen content in the alloy
  - Fluence



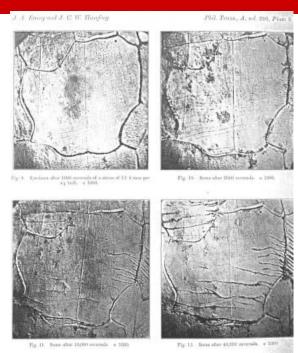
## **Dislocation Channels**

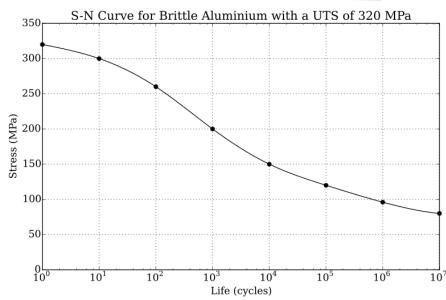
- Channel formation is an important behavior in irradiated zircaloy
- Dislocation channel deformation occurs at high load or high fluence



## **Fatigue**

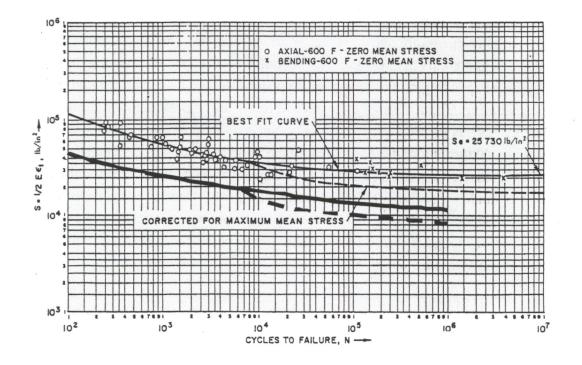
- Material fatigue is the weakening of a material caused by repeated applied loads
- Materials under cyclic loading can experience brittle like fracture at stresses significantly below their UTS
- It is caused by the slow propagation of microstructure damage
- The engineering tool for investigating fatigue is the S-N curve
- An S-N curve is a graph of the magnitude of the cyclic stress (S) against the logarithmic scale of cycles to failure (N).





# **Zr Fatigue**

- Zircaloy experiences fatigue like other metals
- Zircaloy has a fatigue limit at about 10<sup>5</sup> cycles with 180 Mpa (The UTS is >500 Mpa)
- Irradiation slightly lowers the fatigue life in the low cycle range, less than about 10<sup>4</sup> cycles
- Irradiation has no effect on the fatigue life on the high cycle range, greater than about 10<sup>5</sup> cycles
- Design must demonstrate a factor of safety of 2 on stress and 20 on cycles

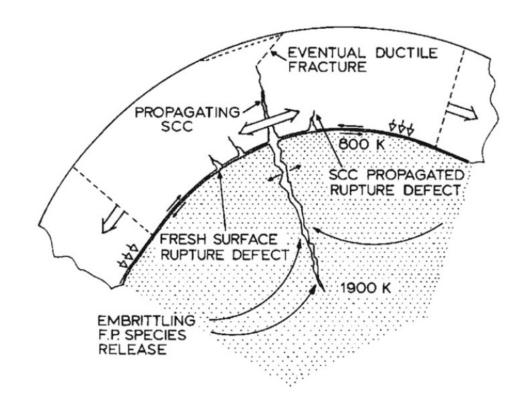


# **Pellet Cladding Interaction (PCI)**

- PCI includes both mechanical (PCMI) and chemical interaction (PCCI), often seen in a more general sense FCMI and FCCI
- The process of PCI involves the combination of high internal mechanical stress in the cladding and a corrosive environment resulting from the accumulation of volatile fission products in the fuel-clad gap
- These conditions may lead to the initiation and propagation of radial cracks, which can penetrate through the cladding
- Frequent PCI failures were observed in the 1960s and 1970s, leading to extensive research efforts
- Altering operational procedures, modifying the manufacturing process of the cladding, adding
  a composite layer and/or applying a protective coating to the inner cladding surface can help
  mitigate these failures

# Physical Phenomena in PCI

- The fuel thermally expands and swells
- The cladding undergoes thermal and irradiationinduced creep
- Mechanical interaction between the fuel pellet and cladding generates radial compressive stresses in addition to tensile hoop stresses
- Volatile fission product gases are released into the gap
- The cracking of the fuel pellet together with nonuniform thermal expansion resulting in preferential contact at the pellet interfaces increases local mechanical stresses in the cladding



# **Stress Corrosion Cracking (SCC)**

- Alloys that are in a corrosive environment may develop cracks that would not otherwise develop in the absence of either stress or the corrosive environment – stress corrosion cracking (SCC)
- Stress corrosion cracks may propagate along intergranular (IG) and/or transgranular (TG) paths
- The propagation modes of these cracks may depend on the localized mechanical and chemical conditions that lead to SCC
- In order for SCC to initiate and propagate in any material, four conditions are simultaneously required:
  - A corrosive environment, a susceptible material, sufficient stress, and sufficient time