

1) 625 K for 400 days $t_e = 500 \text{ mm}$ Vaughn

$$\delta^* = 5.1 e^{-\frac{550}{625}} = \cancel{12.3} \quad 2.115 \text{ } \mu\text{m} \quad \checkmark$$

$$t^* = 6.62 \cdot 10^7 \cdot e^{\frac{11949}{625}} = 133 \text{ days} \quad \checkmark$$

15/18

$$\delta = \delta^* + K_L(t - t^*) \quad \checkmark$$

$$K_L = 7.48 \cdot 10^{-6} e^{-\frac{12500}{T}} = 1.542 \cdot 10^{-11} \frac{\mu\text{m}}{\text{day}}$$

Instead I'll approximate $K_L \approx \frac{\delta^*}{t^*}$
because of cyclic nature of corrosion, linear regime is effectively many cubic regimes so

$$K_L = 0.0159 \frac{\mu\text{m}}{\text{day}} \quad \star \text{ Far more realistic}$$

$$\delta = \delta^* + K_L(t - t^*) = 2.115 + 0.0159(400 - 133)$$

$$= \boxed{6.361 \text{ } \mu\text{m} = \delta(400 \text{ days})}$$

see other sheet for corrected equation
 $\delta = 6.2315 \text{ } \mu\text{m}$

b) $f = 0.18$ $P_{Zr} = 6.5$ $P_{ZrO_2} = 5.68$ $PBR = 1.56$

Use equation from class

time = 365 dy, update δ

$$C_H^{\text{clad}} = \frac{2F \cdot \delta \cdot P_{ZrO_2} \cdot F_{ZrO_2}^0 \cdot \frac{M_H}{M_O}}{(t_{\text{clad}} - \frac{\delta}{PBR}) P_{Zr}} \cdot 10^6 = \frac{2 \cdot 0.18 \cdot 6.361 \cdot 5.68 \cdot \frac{2}{3} \cdot \frac{1}{16} \cdot 10^6}{(500 - \frac{6.361}{1.56}) 6.5}$$

$$= 1.68128 \cdot 10^{-4} \cdot 10^6 = \boxed{168.13 \text{ Wt. PPM} = C_H^{\text{clad}}}$$

From Olander, Matter

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seems low, I probably have a mistake on my equation or entered it in my calculator wrong

2) 3/4

Inward diffusion of O_2 is the rate limiting step
- through oxide layer

3)

The PB ratio measures the effective volume occupied by a metal atom in an oxide against the volume occupied by the atom in matrix.

~~For Zr, a For a metal M, if its PB ratio~~

So for Zr, we have the unit cell volume V_{Zr} divided by the number of Zr atoms per unit cell N_{Zr} , if the same factors are defined for zirconia the PB ratio is:

$$\frac{\frac{V_{ZrO_2}}{N_{ZrO_2}}}{\frac{V_{Zr}}{N_{Zr}}} \rightarrow \text{If } PBR \in (1, 2) \text{ the oxide is likely stable and adherent.}$$

4)

Hydrides often form an outer rim in the cladding. This is due to the preferential precipitation of hydrides in regions of high stress and low temperature (i.e. solubility). They can also form/migrate to crack tips as we covered in DHC. The main factor in the rim precipitation is the intense hoop stress on the outer region of the pellet. They are also subject to local corrosion kinetics. If an area has unusually rapid oxidation (possibly due to spallation) there will be more released hydrogen for the cladding to uptake.

- impacts of hydrides? embrittlement

5) A Reactivity Initiated (or Insertion) accident usually occurs most frequently when control rods are unexpectedly removed from the system. In PWR this is often a Control Rod Ejection Accident (CREA) while in BWR it is a Control Rod Drop Accident (CRDA). In both cases a mechanical failure enables either coolant pressure (in CREA) or gravity (in CRDA) to remove the control rod blade from the core. + Pressure pulses are a key concern

The loss of control rod causes a sharp increase in core temperature (~0.1 s) this can lead to several negative outcomes* starting with rapid fuel swelling and consequent PCMI, then as well as cladding and hydride failure in cladding. It can also cause departure from nucleate boiling, ballooning, fuel fragmentation and release.

6) A Loss of Coolant Accident occurs when the flow of coolant _{to the core} is reduced or lost altogether.

The timescales for LOCAs are much longer and they occur on the order of tens of seconds to minutes. Other than the timescales, LOCA and RIAs have different safety concerns.

In LOCA, superheated steam poses a major oxidation risk. Runaway oxidation, as well as quenching from ECCS are major concerns.

Oxidation is particularly dangerous due to the buildup of H_2 in the core. - got most of it

7)
4/5 One major way burnup impacts type of failure is in cladding failure. Irradiation embrittles Zr cladding making brittle failure more likely in accidents involving high burn-up fuel. This is especially pronounced in LOCA as the typical failure mechanism is severe plastic deformation.
- for A14 also - corrosion + hydrides

8)
8/8 - Improved Corrosion Kinetics ✓
FeCrAl alloys are attractive for their corrosion performance under LOCA conditions ^{together (thin films also look promising)}
- Improved Fuel Properties ✓
Uranium based fuels with additives and dopants might offer better thermal conductivity which may allow for lower operating temperatures
- Improved cladding properties ✓
SiC composite cladding ✓
- Retention of fission products ✓
Fully ceramic microencapsulated fuels ✓

9)
4/6 Zr cladding undergoes rapid oxidation releasing significant amounts of hydrogen. ZrO_2 is less thermodynamically stable in superheated steam which ~~also~~ allows for "break away" corrosion. O_2 dissolves into some of the matrix, forming stabilizing αZr , while some of the cladding will transition to βZr , the combination of α, β can allow for super-plasticity. Even without these phase changes creep is accelerated in Zr at high temperatures.

10)

4/6

- Cladding oxidation and Hydrogen Pickup obviously plays a huge role as a limiting phenomena given it is ~~even~~ deleterious under operating and some accident conditions.

- PCMI ✓

Pellet cladding mechanical interaction poses a high risk factor for failure. Limits focus on hoop strain ~~in the~~ and keeping it below 1%

- Power to Melt ✓

Seems like a well met limit with current fuels the powers needed to melt are much higher than operational range $\sim \text{LWR} \approx 600 \frac{\text{W}}{\text{cm}}$

11)

3/6

CND can negatively impact water chemistry and corrosion. It can also sequester boric acid and cause CRIPs. ✓

- under more

12)

~~DH~~

4/6

Dissolved hydrogen is injected into coolant to react with radiolysis products (e.g. OH) so that oxidizing species in the coolant are minimized. ✓

Lithiation ~~controls the~~ is often used to control the pH to ~ 6.9 . This promotes oxide stability and helps mitigate corrosion. ✓

For 1 Vaughn

I have 10^{-6} where it should be 10^6 on my sheet

$$K_L = 0.015417$$

$$\delta = 2.115 + 0.015417(267) = 6.2315 \text{ mm} \quad \checkmark$$

$$K_L = 7.48 \cdot 10^6 \cdot e^{-\frac{12500}{T}} = 0.015417 \quad \checkmark$$

checks out
with $\frac{\delta^*}{\delta^*}$

solution