

NE 591: Advanced Reactor Materials

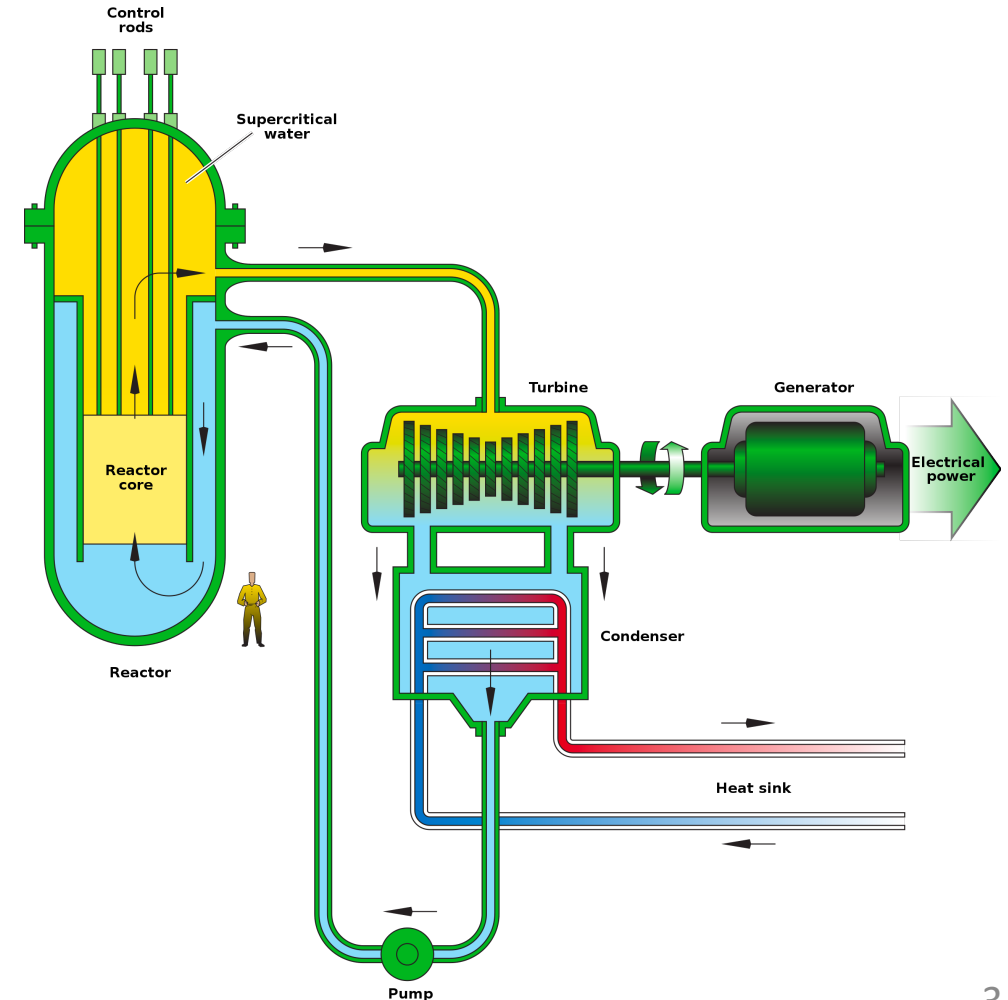
Fall 2021

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SUPERCRITICAL WATER COOLED REACTORS

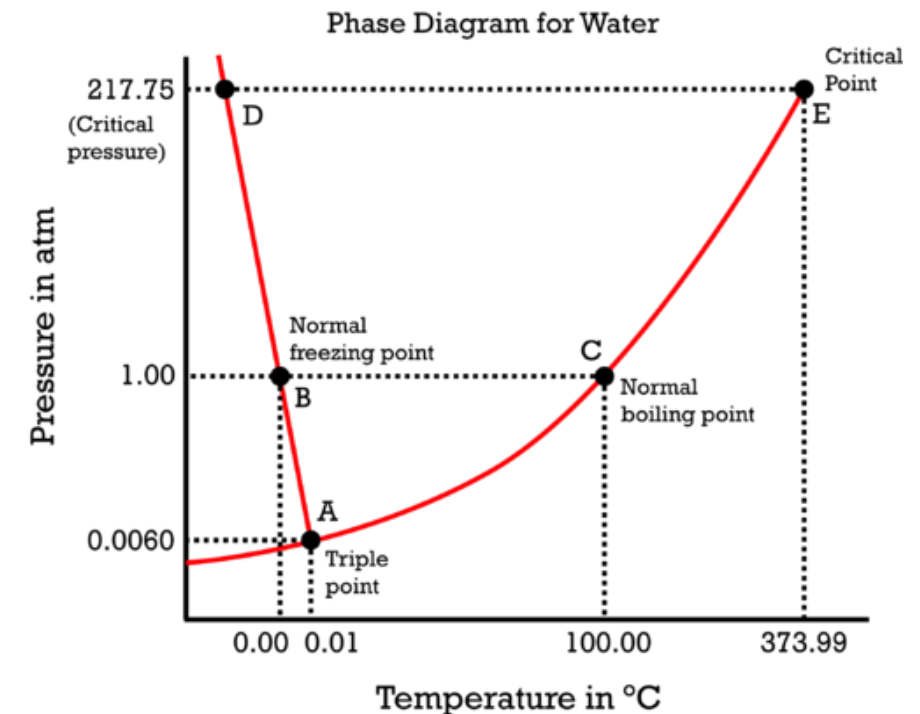
Supercritical Water Reactor (SCWR)

- Similar to a traditional light water reactor, but operates at a supercritical pressure
- Water heated in the core becomes a supercritical fluid, which can directly be used in a steam turbine
- Removes the added step of a heat exchanger, increasing the efficiency of the reactor system



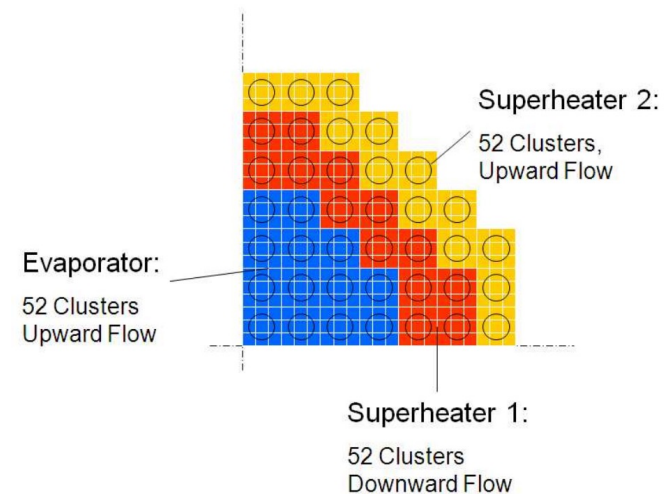
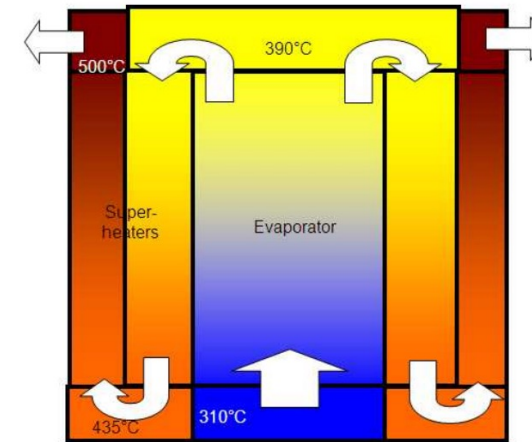
Supercritical water

- A supercritical fluid is a substance with a temperature and pressure above its critical point
- At such a point, distinct solid and liquid phases do not exist
- Critical point of water is 22 MPa and 647 K
- Density is highly variable, based on temperature and pressure:
 - 0.78 g/cc @ 25 MPa and 280 C
 - 0.09 g/cc @ 25 MPa and 500 C
 - these are reasonable inlet/outlet temperatures for SCWR coolant



Comparison to LWR

- SCWR
 - coolant in: 280 C
 - coolant out: 600 C
 - outlet specific enthalpy: 3150 kJ/kg
 - three pass coolant
 - coolant pressure: 25 MPa
- LWR
 - coolant in: 275 C
 - coolant out: 315 C
 - outlet specific enthalpy: 200 kJ/kg
 - one pass coolant
 - coolant pressure: 15 MPa



Features

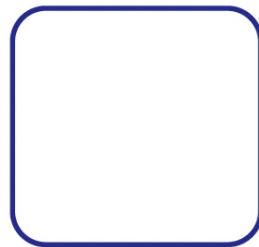
- Supercritical water has excellent heat transfer properties allowing a high power density, a small core, and a small containment structure
- The use of a supercritical Rankine cycle with its typically higher temperatures improves efficiency (would be ~45 % versus ~33 % of current PWR/BWRs)
- This higher efficiency would lead to better fuel economy and a lighter fuel load, lessening residual (decay) heat
- SCWR is typically designed as a direct-cycle, whereby steam or hot supercritical water from the core is used directly in a steam turbine with no intermediate heat exchangers, etc
- SCWRs can operate as a fast breeder, or utilize heavy water and the thorium fuel cycle
- Disadvantages include operating at very high temperatures and pressures, increasing material challenges
- The economic advantage of a direct cycle is a downside with regard to safety, in that a cladding breach means your turbine and generators are directly exposed to radioactivity
- Corrosion in SCW is a challenge

SCW geometry

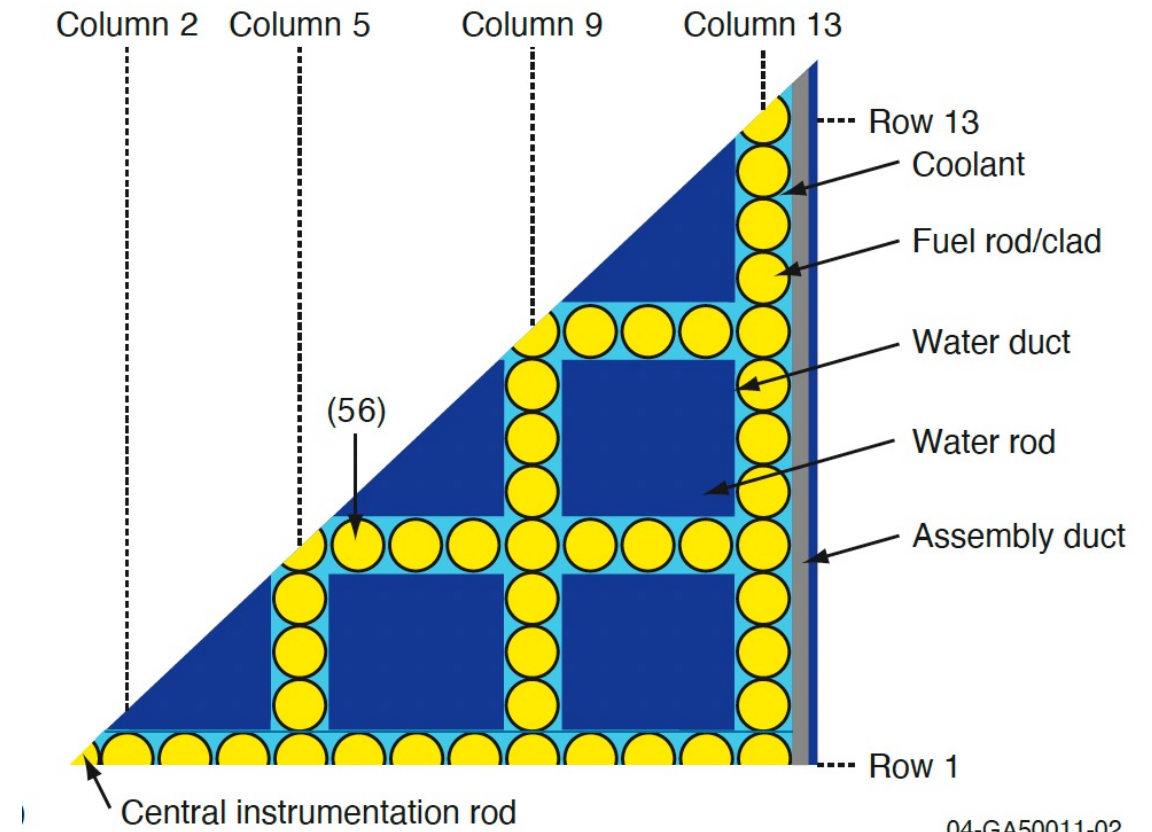
- There is experience in SCW cycles in the fossil fuel industry, however they are able to use thick-walled geometries that are not feasible in nuclear designs



Fuel rod outer diameter = 12 mm
Clad wall thickness = 0.4–0.6 mm



Water rod outer diameter = 40 mm (square)
Water rod wall thickness = 0.4 mm



04-GA50011-02

1/8 assembly model SCWR core

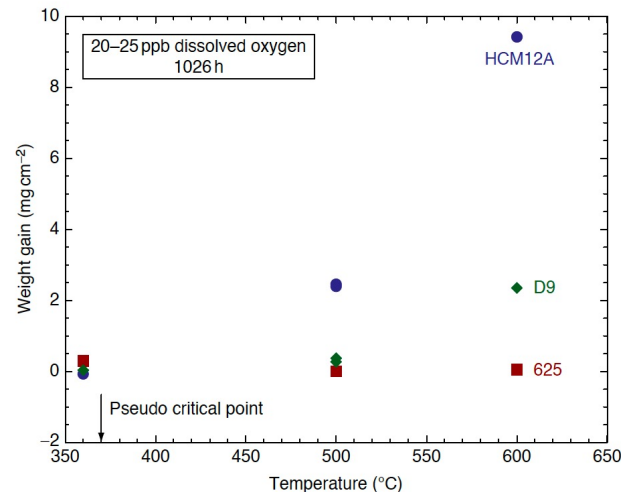
Corrosion

- These very thin-walled components provide little margin for corrosion in an SCWR core, where the consequences of failure are significant
- Oxide films and deposition of corrosion products from out-of-core components can lead to overheating (and failure) and changes in reactivity
- SCW corrosion studies have included F/M steels, austenitic steels, Ni alloys, Zr alloys and Ti alloys

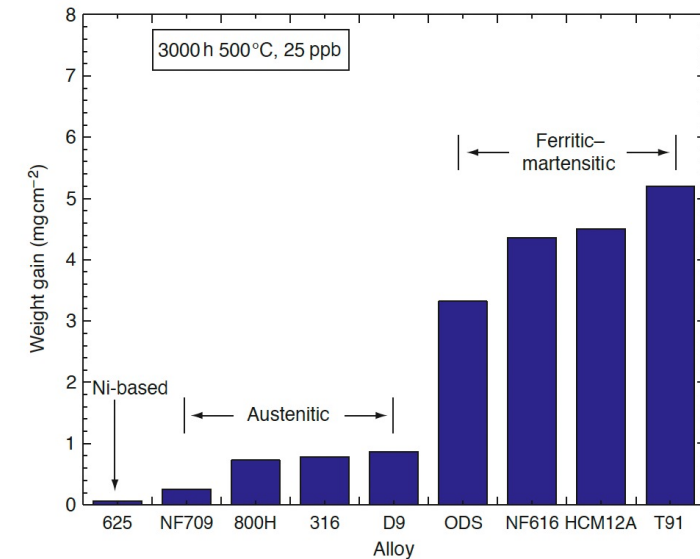
<i>Alloy class</i>	<i>Alloy</i>	<i>Temp. (°C)</i>	<i>Water chemistry</i>	<i>Exposure time (h)</i>
Austenitic stainless steel	304, 304L, 316, 316L, 316 + Zr, 310, 310S, 310 + Zr, 347H, Sanicro28, D9, 800H, AL6XN, Carpenter 20C B3, Nitronic-50, PNC1520, alloy 1.4970	290–650	Deaerated (< 10 ppb) to 8000 ppb dissolved oxygen	100–3000
Nickel-based	600, 625, 690, 718, 825, C22, B2, C276, MAT21, MC	290–600	Deaerated (< 10 ppb) to 8000 ppb dissolved oxygen, < 0.1 mS cm ⁻¹	100–3000
Ferritic-martensitic	T91, T91a, T91b, HCM12A (T122), HCM12, HT-9 (12Cr–1Mo–1WVNb), NF616 (T92), MA956, 2.25Cr–1Mo (T11), P2	290–650	Deaerated (< 10 ppb) to 8000 ppb dissolved oxygen, < 0.1 mS cm ⁻¹	100–3000
Oxide dispersion strengthened	9Cr, 12Cr, F/M, 316, Inconel, Hastelloy G-30, 19Cr, 14Cr–4Al, 16Cr–4Al, 19Cr–4Al, 22Cr–4Al	360–600	25 ppb	200–3000
Zirconium-based	Zr, Zr–Nb, Zr–Fe–Cr, Zr–Cr–Fe, Zr–Cu–Mo, Zr-2, Zr-4	400–500	Deaerated (< 10 ppb dissolved oxygen), < 0.1 mS cm ⁻¹	< 2880
Titanium-based	Ti–3Al–2.5V, Ti–6Al–4V, Ti–15Mo–5Zr–3Al, Ti–15V–3Al–3Sn–3Cr	290–550	8000 ppb dissolved oxygen, 0.1 mS cm ⁻¹	500

Alloy Classes

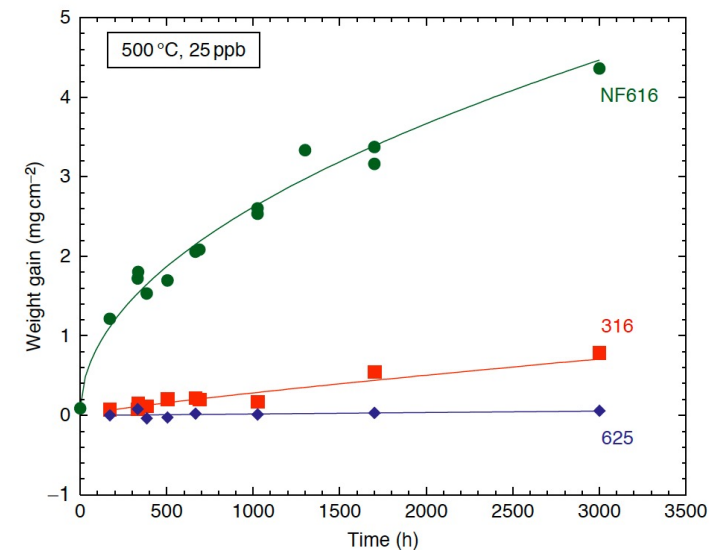
- Corrosion is strongly dependent upon the alloy class
- Oxidation is most rapid for F/M steels and slowest for Ni alloys
- Temperature has a very strong effect on corrosion



HCM12A = F/M
D9 = aust.
625 = Ni



Dissolved
O content
25 ppb



NF616 = F/M
316 = aust.
625 = Ni

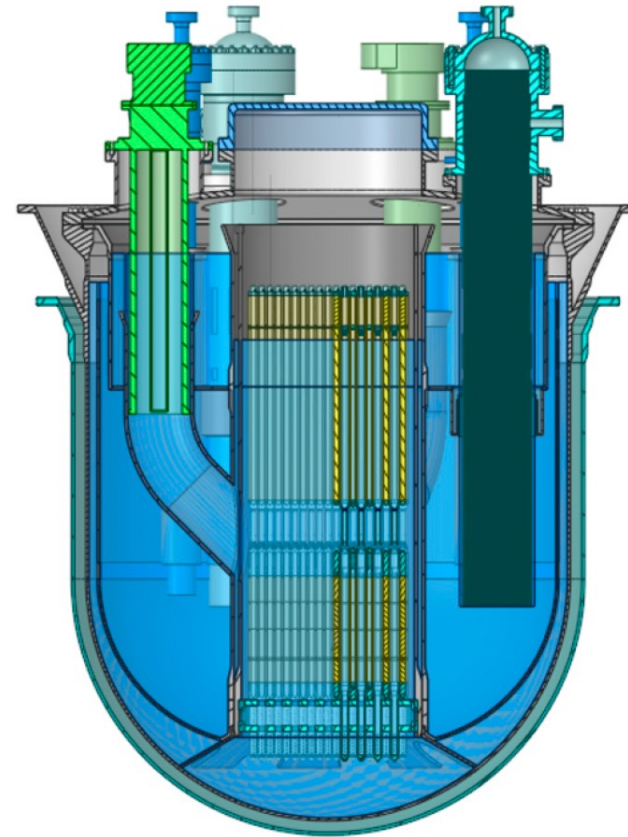
Stress Corrosion Cracking

- Austenitic stainless steels and nickel-based alloys exhibit susceptibility to IGSCC in pure SCW over the temperature range of 400–650C
- IGSCC decreases with temperature, but overall fracture increases with temperature
- Small additions of HCl or H₂SO₄ increase susceptibility to IGSCC in austenitic alloys
- Ni based alloys are more susceptible to SCC than austenitic steels
- Higher Cr content in Ni alloys seems to reduce the extent of SCC
- F/M alloys are generally resistant to SCC (HT-9 is an exception)
- Irradiation strongly exacerbates SCC, but the effect decreases with increasing temperature

LEAD COOLED REACTORS

ALFRED Reactor

- ALFRED, the advanced lead-cooled fast reactor European demonstrator
- ALFRED is today a demonstrator reactor designed with the specific purpose to test and qualify innovative components and procedures to be used in commercial reactors
- Pool type, MOX hollowed fuel pellets, natural and forced circulation
- Planned commercial operation in 2035-2040



ALFRED Reactor

- ALFRED plans to operate in a staged manner for licensing purposes, ramping up activities
- Currently planned materials are listed for the different stages, relying heavily on coatings and austenitic steels
- An example 15-15Ti steel in the US is D9
- Different reactor concepts with different candidate materials

Table 2 ALFRED main parameters in the stages of operation

	Stage 0 (Commissioning)	Stage 1 (Low temp.)	Stage 2 (Medium temp.)	Stage 3 (High temp.)
Core inlet temperature (°C)	390	390	400	400
Core outlet temperature (°C)	390	430	480	520
Core thermal power (MW)	≈ 0	100	200	300

Table 3 ALFRED RCS SSCs candidate materials and protective measures

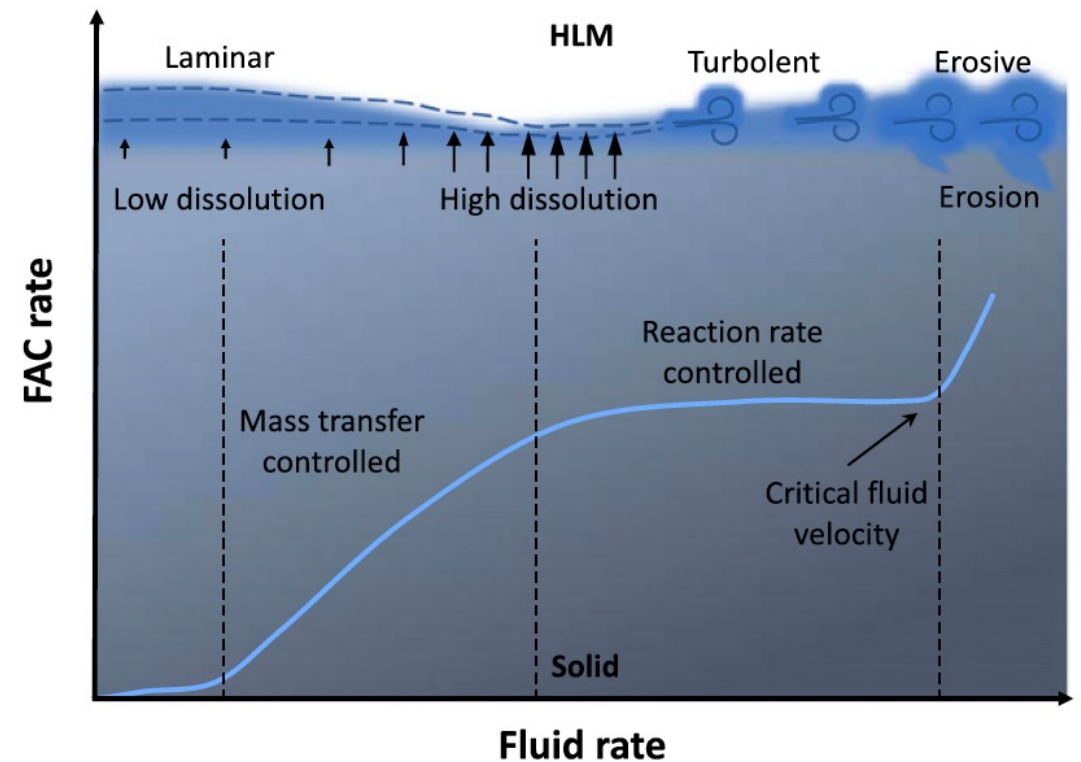
Component/Part	Stage 1	Stage 2	Stage 3
FA/cladding	15-15Ti (AIM-1)	15-15Ti (AIM-1) + PLD Al ₂ O ₃ coating	15-15Ti (AIM-1) + PLD Al ₂ O ₃ coating
IV	AISI316L ^a + Al ₂ O ₃ coating ^b	AISI316L ^a + Al ₂ O ₃ coating ^b	AISI316L ^a + Al ₂ O ₃ coating ^b
IS	AISI316L ^a + Al ₂ O ₃ coating ^{b,c}	AISI316L ^a + Al ₂ O ₃ coating ^{b,c}	AISI316L ^a + Al ₂ O ₃ coating ^{b,c}
HEX	AISI316L ^a	AISI316L ^{a,d}	AISI316L or T91 + Al ₂ O ₃ coating ^b , or AFA steel ^e
RCP/shaft, impeller	AISI316L + Al ₂ O ₃ coating ^b	AISI316L + Al ₂ O ₃ coating ^b	AISI316L + Al ₂ O ₃ coating ^b
RV	AISI316L ^a	AISI316L ^a	AISI316L ^a

Lead Corrosion

- The challenge of using heavy liquid metal coolant (lead and lead-bismuth) is the compatibility with structural materials at high temperature
- Pb and lead-bismuth eutectic (LBE) provide highly corrosion conditions for most structural and cladding materials
- This is most problematic for thin-walled components such as cladding tubes or heat exchangers
- Corrosion in Pb and LBE is due to the dissolution of different constituent elements, their transport, and their chemical reactions with oxygen and nitrogen

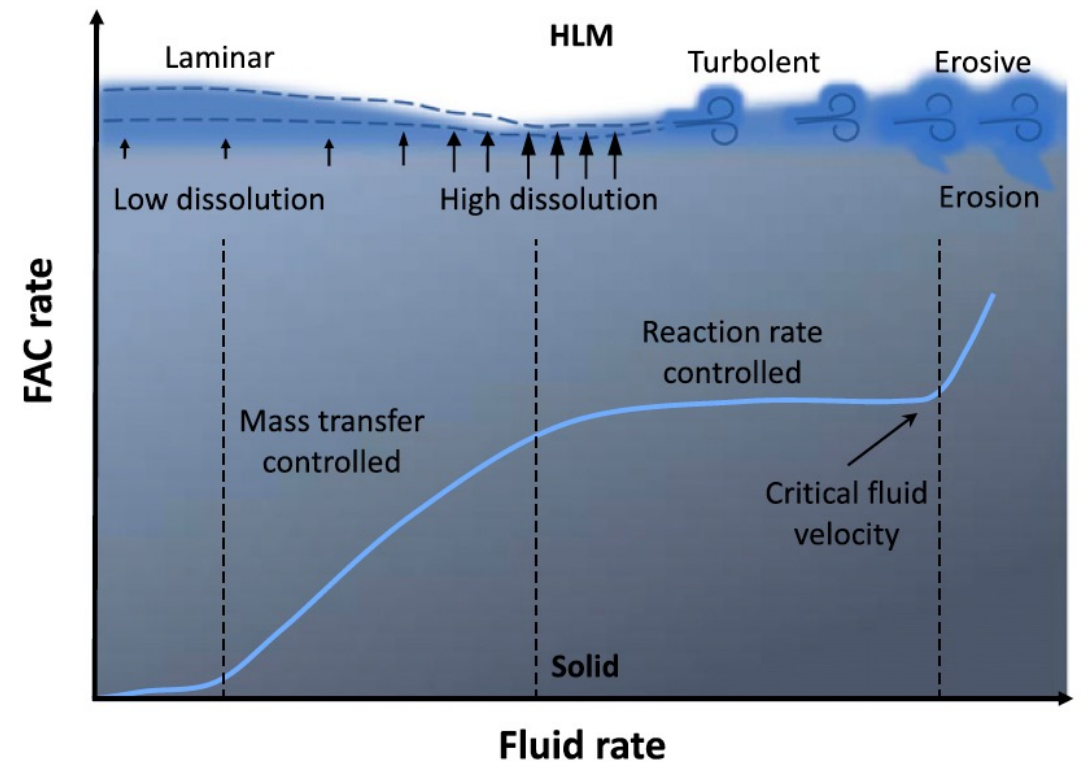
Flow Accelerated Corrosion

- At low flow velocity, the corrosion is controlled by mass transfer, that is by diffusion of dissolved elements in the boundary layer
- When the flow velocity increases, the thickness of the laminar boundary layer decreases, leading to an increase of the corrosion rate; the enhanced flow velocity locally reduces the corrosion product concentration at the interface, further promoting the diffusion process in the boundary layer



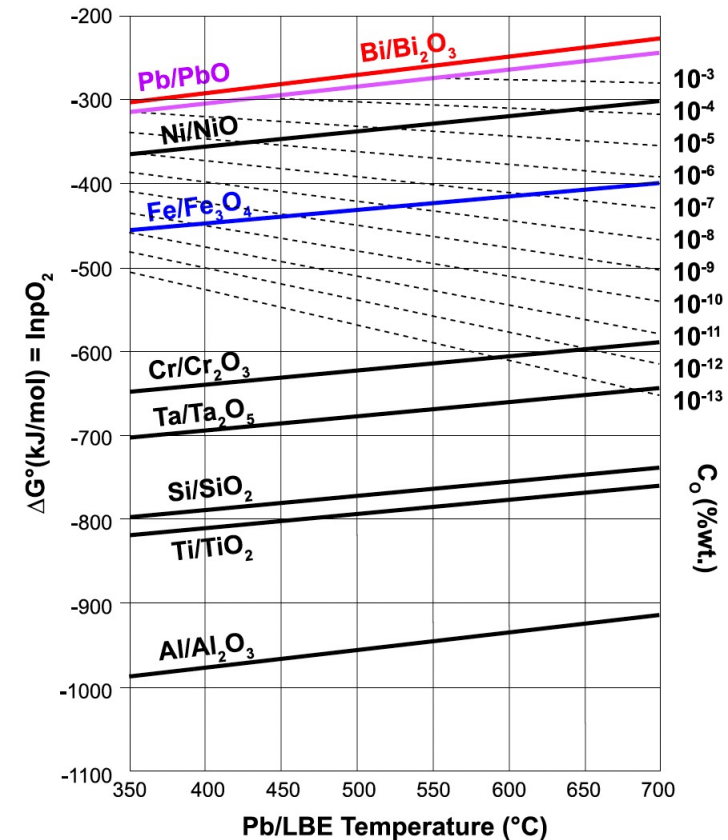
Flow Accelerated Corrosion

- When the velocity is high enough to “clean” the surface of the solid metal from the corrosion products, the corrosion becomes flow-independent and therefore it is considered reaction rate controlled
- At higher flow velocities, above a critical value, erosion-corrosion and also cavitation-corrosion may occur, leading to a significant increase of the corrosion rate



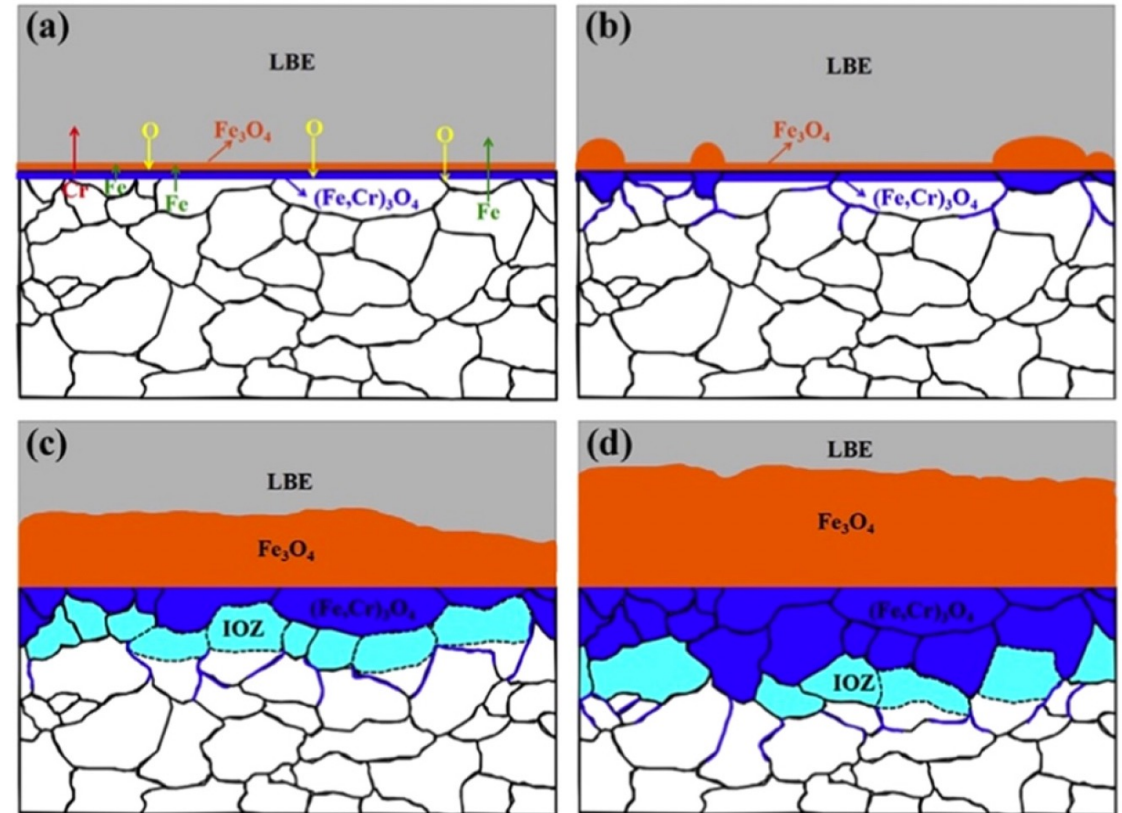
Impurities

- A factor that has a considerable effect on the corrosion rate processes is the presence of nonmetallic impurities such as carbon, nitrogen, and oxygen
- Impurity elements can take part in corrosion reactions and can form compounds with dissolved species
- Oxides of the steel components are more stable than the LBE, and thus a Fe-Cr oxide layer on steel surfaces



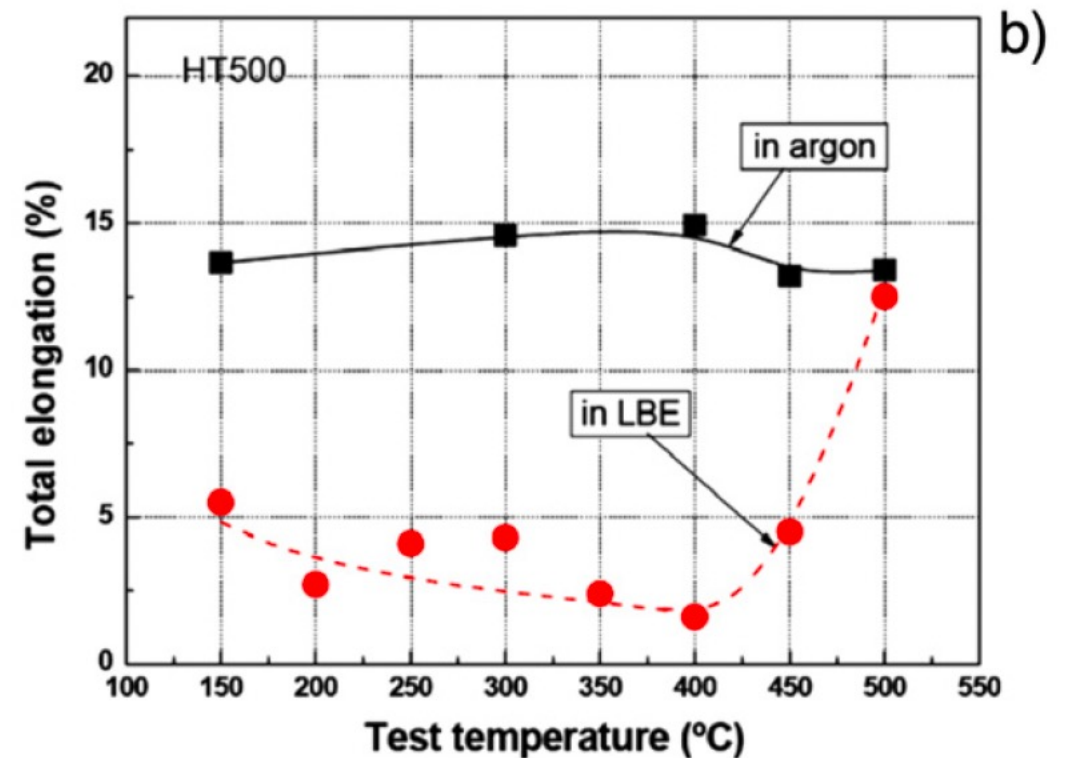
Oxide Passivation

- Adequate control of the oxygen concentration allows one to maintain the Fe-Cr oxide film, without oxidizing the LBE
- If insufficient oxygen is present, the steel will suffer dissolution corrosion
- The oxide layer can form as a multi-layered structure dependent upon O content, temperature, and flow, with Fe_3O_4 and FeCrO_2 phases



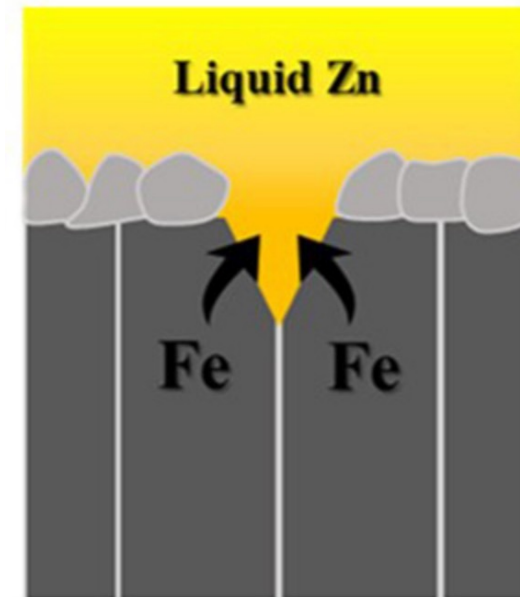
Mechanical Property Effects

- Liquid metal embrittlement (LME) is a known phenomenon, but its mechanism is not fully understood
- Contact with LM causes a decrease in ductility, resulting in a premature brittle failure in an otherwise ductile material
- LME occurs by nucleation of a crack at the wetted surface of a solid and the subsequent propagation into the bulk until ultimate failure



LME

- The loss of ductility of solid metals when in contact with liquid metal appears to be limited to a particular temperature range known as the ductility trough
- For the F/M T91 steel in liquid lead, a ductility trough between 160C and 425C was found
- The ductility trough is detected by following the evolution of the ultimate tensile stress or energy to rupture with temperature
- The working theory on LME is based on a reduction of the surface energy by absorption of liquid metal atoms at the crack tip of the solid



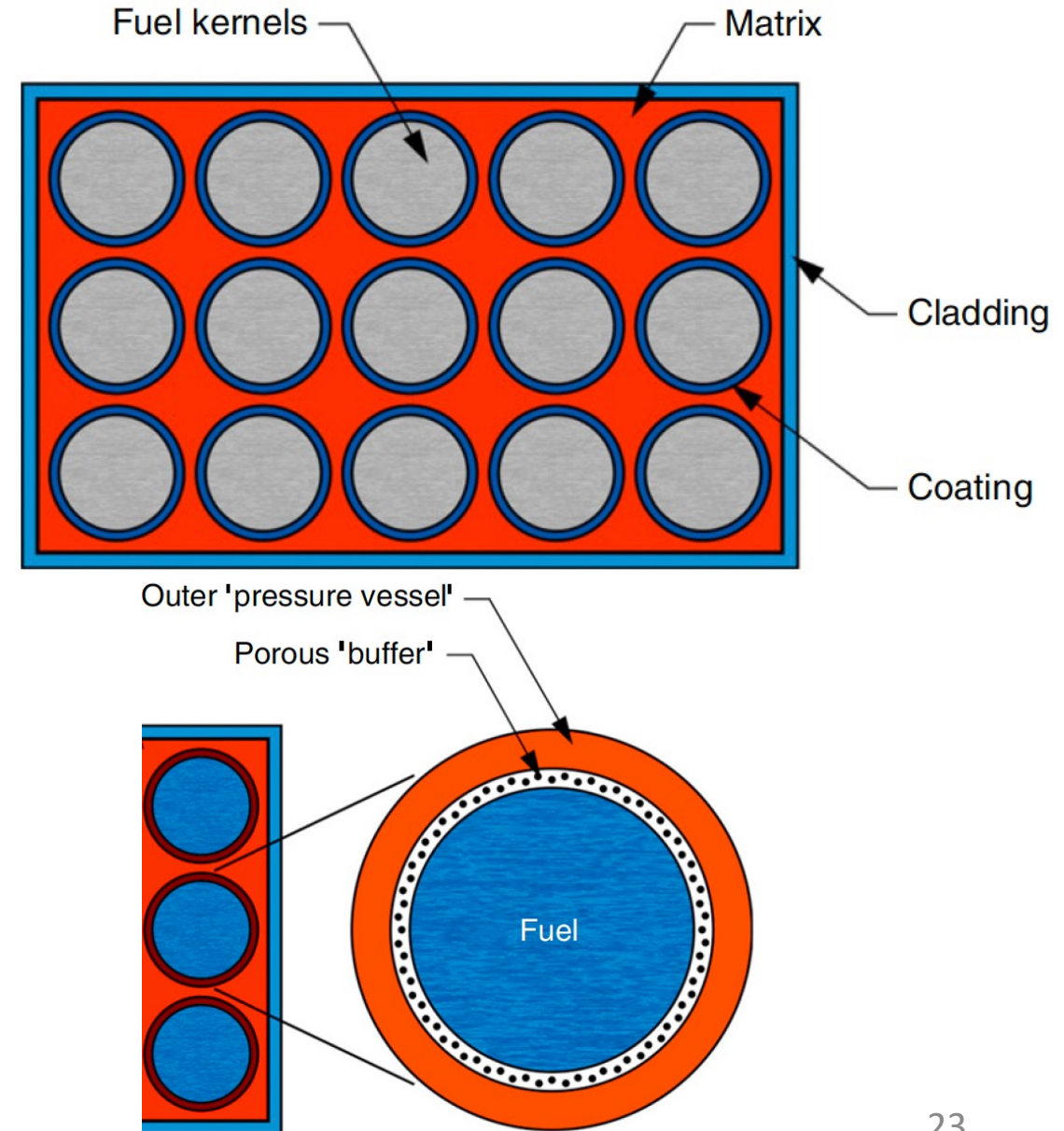
COMPOSITE FUELS

Composite Fuels

- Composite fuels consist of a fissile phase dispersed in an inert, nonfuel matrix
- These types of fuels have been used since the 1950s in high-temperature and high-power density applications
- Composite fuels are typically distinguished into two types: Cermet and Cercer
- Cermet: ceramic fuel particles dispersed in a metallic matrix
- Cercer: ceramic fuel particles in a ceramic matrix
- 47 different fuel matrix combinations have been experimentally explored
- Composite fuels have potential as high burnup fuels to either transmute transuranic elements, or to replace conventional UO₂ fuel pellets

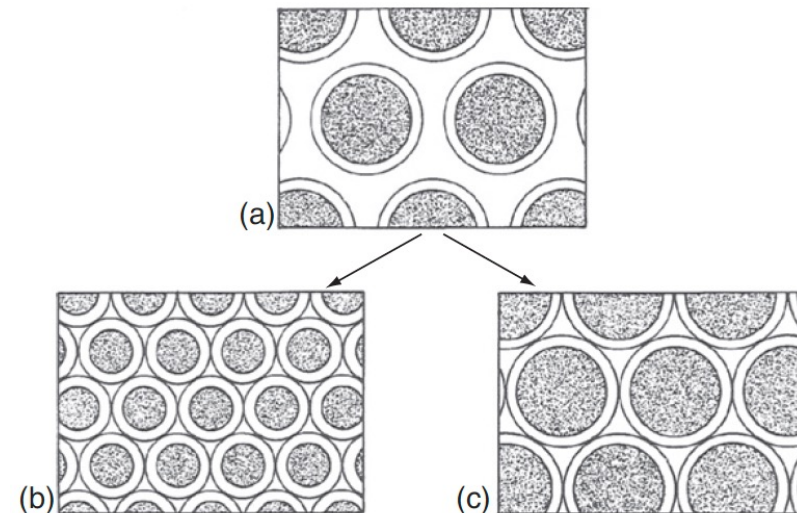
Composite Fuels

- The most complete experimental work focused on UO_2 particles dispersed in stainless steel
- The primary feature that distinguishes composite fuels from the pellet-in-cladding fuel types is the localization of fuel material within an inert matrix
- Fuel particles can be individually clad within the matrix material by an individual coating



Cermet Fuel

- The performance of the Cermet fuel depends on the ability of the matrix to retain strength and ductility during irradiation
- The strength of the matrix must be sufficient to resist cracking from stresses due to solid fission products and fission gas swelling
- The major source of matrix degradation is the fission fragments ejected from the fuel
- The distance between (or density) of fuel particles must be sufficient such that the damage region around a given particle does not overlap with an adjacent particle



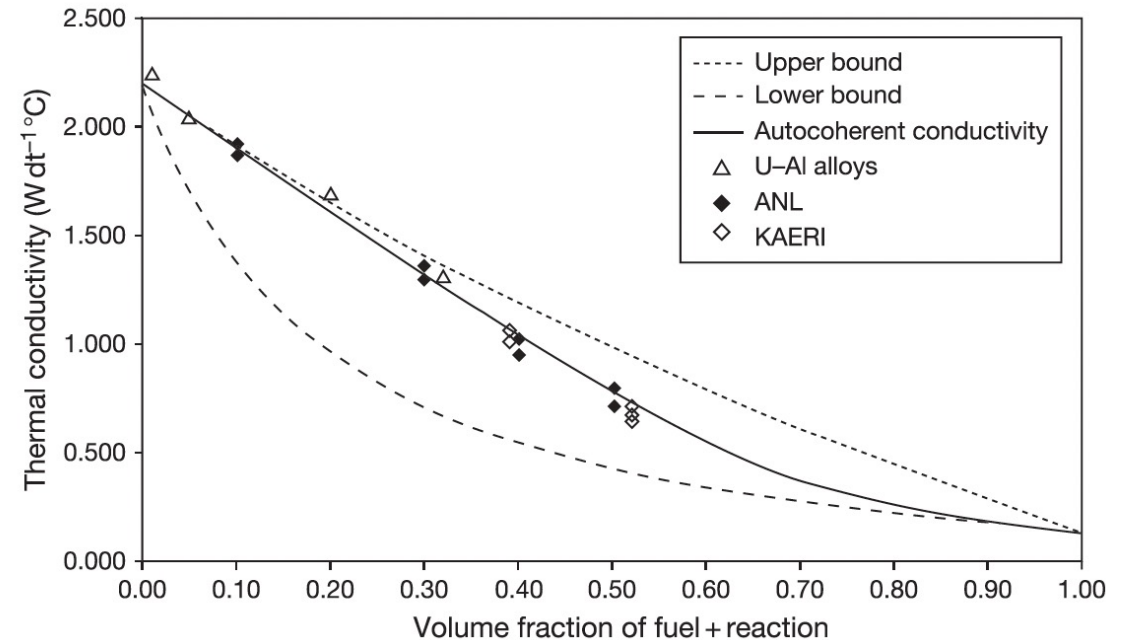
Cercer Fuel

- The typical failure mode in cercer fuels is fuel particle swelling that leads to a tensile stress in the matrix, followed by matrix cracking
- Models have been constructed to predict failure taken from thermal expansion in duplex ceramics
- Increasing the interparticle distance, decreasing the particle size, decreasing fuel swelling and thermal expansion, etc., reduce the probability of fracture of the matrix

$$\left[\frac{\beta R_p (\beta + 2)}{\pi (\beta + 1)} \right]^{1/2} \left[\left(\frac{1}{(\beta + 1)^2} + 6 \frac{\sqrt{2}}{\pi} V_p \right) c_4 \right] > K_{IC}$$

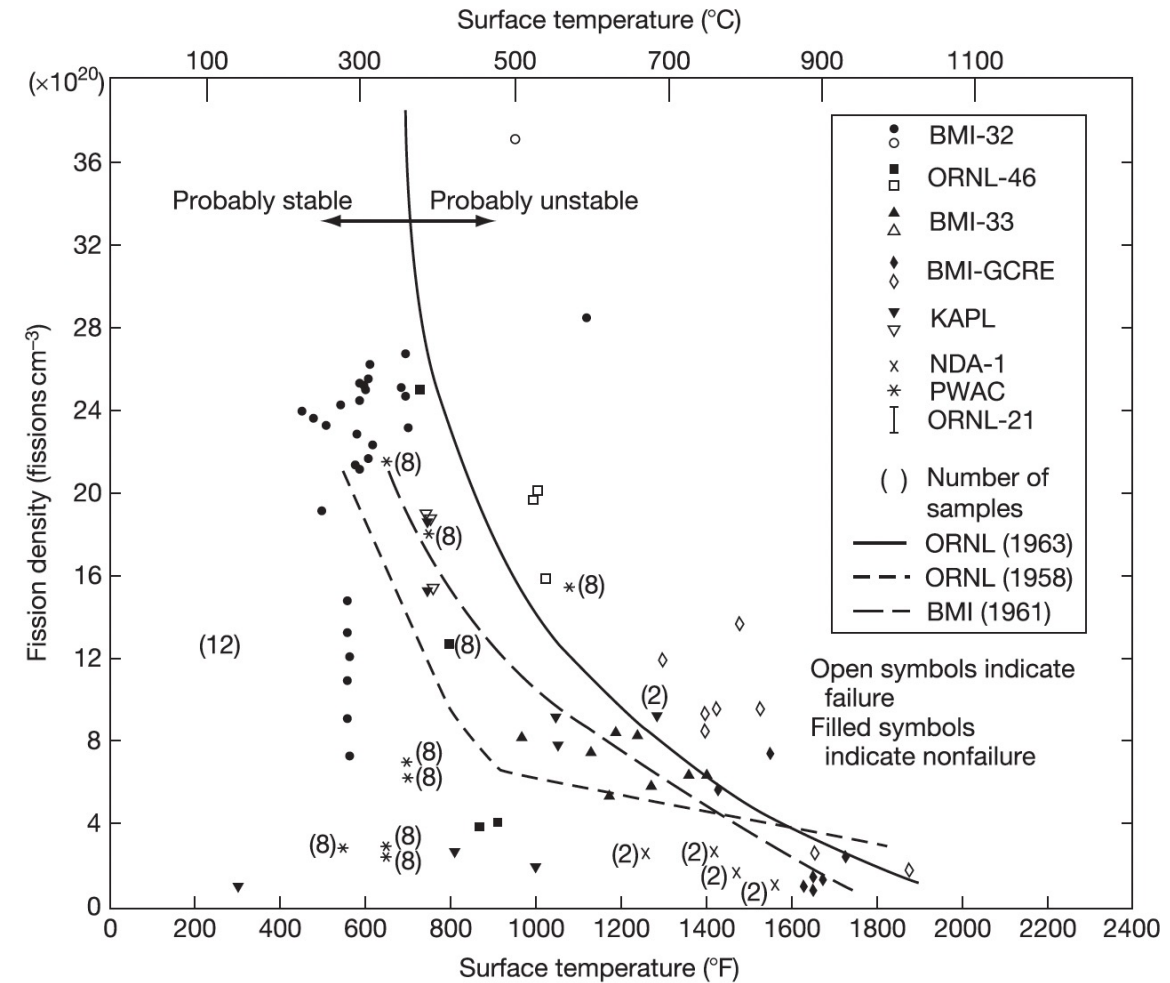
Composite Thermal Conductivity

- The behavior of composite fuels can be heavily influenced by fuel operating temperature, and thermal conductivity is of primary importance in the irradiation behavior of these fuels
- Fuel particle size, shape, orientation, and distribution, and matrix porosity and the thermal conductivities of the fuel and matrix species are the key properties of interest



Irradiation Behavior

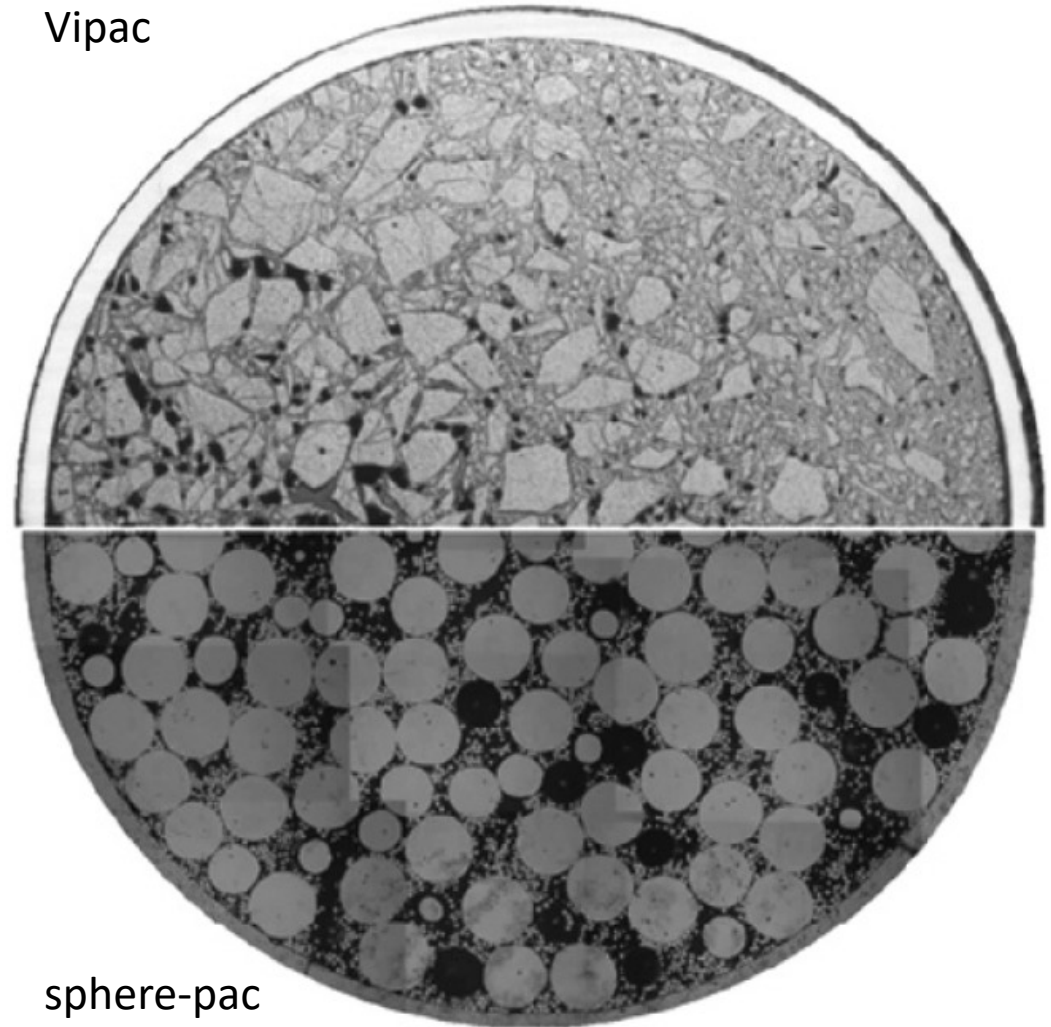
- Most irradiation information is on UO₂ dispersed in stainless steel
- Full-size 20vol% plate-type elements reached 65 at% burnup at surface temperatures of 315–427C with no blistering or cracking
- There are bounding limits on fission density (+ loading) and surface temperature
- Higher matrix loadings are possible, but are restricted to low (<15%) burnups



PARTICLE FUELS

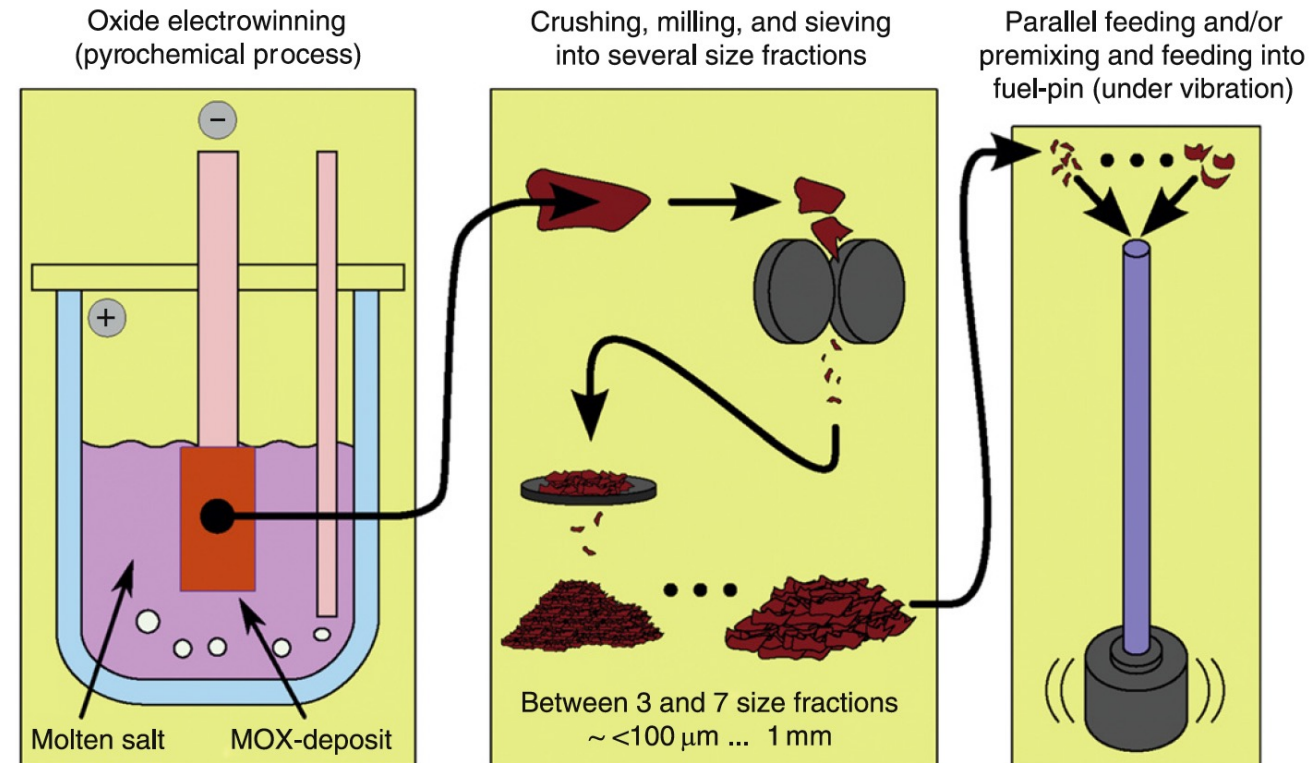
Particle Fuels

- When the fissile isotopes are coming from spent fuel that is chemically separated (reprocessed), particle fuel with its direct filling of fuel particles into the fuel pin offers several advantages
- Two major types of particle fuel are sphere-pac and Vipac



Vipac

- The usage of randomly shaped (angular shards) particles as nuclear fuel component filled in a cladding goes back to the 1950s
- Reprocessing of fuel from the BOR-60 reactor was pursued to develop fuel for the BN-600 reactor – MOX fueled fast reactors
- These particles are fed into the cladding under vibration
- Vibration packing = Vipac



Sphere-Pac

- Sphere-pac fuel is also composed of particles which are directly filled into a fuel pin
- However, particles are shaped into spheres to generate a more predictable arrangement of particles in the fuel pin
- Sphere-pac is used in conjunction with the PUREX process, combined with follow-up processes such as actinide extraction
- A fully aqueous method offers excellent distribution of the fissile material in the matrix and the formation of solid solution in the ceramic
- This also eliminates the need for mechanical devices, reduces powder generation, and thus minimizes contamination

Particle Fuels

- This modified geometry of fuel will present important changes in the fuel properties as a unit
- The thermal conductivity is reduced, which increases the centerline temperature, and can lead to restructuring
- The fuel is softer because of the void space, reducing FCMI
- Significantly higher fission gas release, due to the prevalence of free surfaces
- In the case of a cladding breach, this increased fuel surface area provides greater opportunity for coolant reaction with the fuel
- At fuel startup, fuel particles are not bonded, and in case of early life cladding breach, potential loss of fuel particles