# **Nuclear Fuel Performance**

NE-533

Spring 2023

## **Last Time**

- Pellet-clad interaction (PCI) takes two forms
  - Pellet-clad chemical interaction, PCCI (bonding occurs)
  - Pellet-clad mechanical interaction, PCMI (pellet pushes and drags cladding)
- In order for SCC to initiate and propagate in any material, four conditions are simultaneously required:
  - A corrosive environment, a susceptible material, sufficient stress, and sufficient time
- Two types of mitigation strategies to limit PCI failures

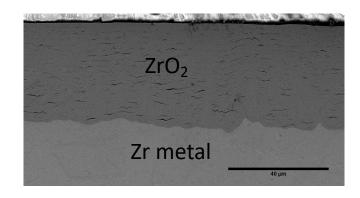
# **CLADDING OXIDATION**

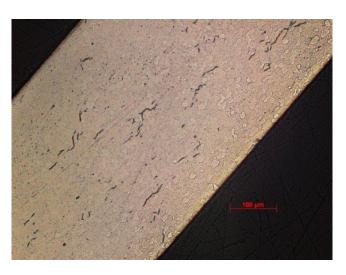
## **Zirconium Oxidation**

- Corrosion of zirconium due to the water coolant is one of the largest concerns for LWR fuel operation
- Corrosion of zirconium alloys in an aqueous environment is principally related to the oxidation of the zirconium by the oxygen in the coolant, dissolved or produced by radiolysis of water

$$Zr + 2H_2O \rightarrow ZrO_2 + 2H_2$$

- Zirconium interacts with water to make zirconia and hydrogen
- Both the oxide layer and the hydrogen adversely impact cladding performance





## Corrosion

- Corrosion is the degradation of a material and its properties under the action of the surrounding environment by chemical and/or electrochemical reaction
- Corrosion reactions require the following:
  - Oxidant

$$Zr + 2H_2O \rightarrow ZrO_2 + 2H_2$$

- Electric current
- A corrosion reaction is composed of an oxidation reaction and a reduction reaction
- During corrosion at a particular spot on the surface of the object, oxidation takes
  place, and that spot behaves as an anode
- The electrons released at this anodic spot move through the metal and go to another spot and reduce H, which behaves as a cathode

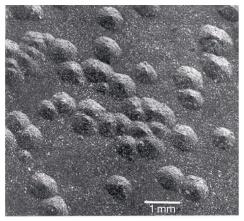
## Corrosion

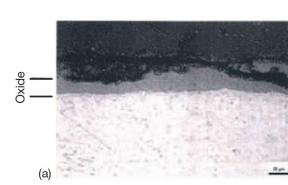
- Corrosion reaction rates are determined by thermodynamics and kinetics
- Thermodynamics tell us whether a material may corrode
  - Measure voltage difference between anodic and cathodic sites
- Kinetics tell us how quickly a material will corrode
  - Measure net current between anodic and cathodic sites

# **Corrosion Types in LWRs**

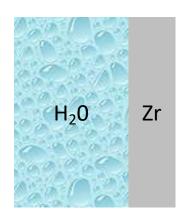
- Several types of corrosion morphologies have been observed in nuclear reactors and in autoclave experiments, of which the most important are: 1) uniform, 2) nodular, and 3) shadow
- Uniform: The formation of a thin uniform layer of zirconium dioxide on the surface of a zirconium alloy component
- Nodular: The formation of local, small, circular zirconium oxide blisters
- Shadow: The formation of local corrosion regions that mirror the shape (suggestive of a shadow) of other nearby noble reactor core components







### Formation of Oxide Layer



#### Oxidation reaction:

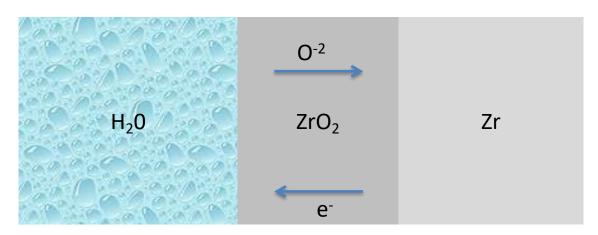
$$Zr(s) \rightarrow Zr^{4+}(aq) + 4e^{-}$$

$$Zr(aq)^{4+} + 2H_2O(l) \rightarrow ZrO_2(s) + 4H^+(aq)$$

#### Reduction reaction:

$$4e^- + 4H^+(aq) \to 2H_2$$

### **After Oxide Formation**



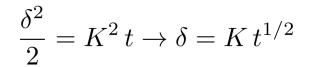
$$2H_2O(1) \rightarrow 2O(aq)^{-2} + 4H^+(aq) \quad Zr^{4+} + 2O^{-2} \rightarrow ZrO_2 + 4e^-$$

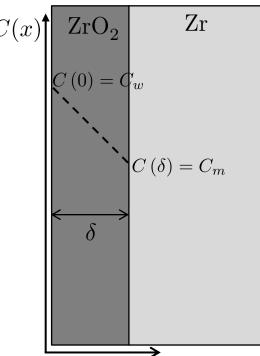
$$4e^- + 4H^+(aq) \to 2H_2$$

- Dissociation of water at oxide/water interface
- 2. Absorption of oxygen into oxide layer
- 3. Diffusion of oxygen through oxide layer
- 4. Reaction of oxygen with zirconium
- 5. Diffusion of electrons through oxide layer
- 6. Reduction of hydrogen

# Corrosion rate limited by diffusion

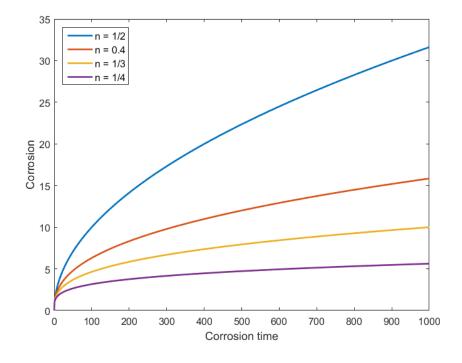
- Because the rate limiting steps are diffusion, we can model the oxidation rate using diffusion
- Assumptions:
  - Transport of O species is rate-limiting
  - Transport of charged species by diffusion only
  - Homogeneous oxide layer
  - No sources/sinks of ions in oxide
  - All oxygen is used to create oxide
  - No loss of oxide





## Observed kinetics are slower than parabolic

- Parabolic kinetics
  - Diffusion of species across the oxide
- Sub-parabolic kinetics
  - Additional ions in oxide
  - Non-uniform electric field in oxide layer
- Once a nonhomogeneous electric field arises, the corrosion rate decreases to the sub-parabolic rate due to inhibition of transport of charged species

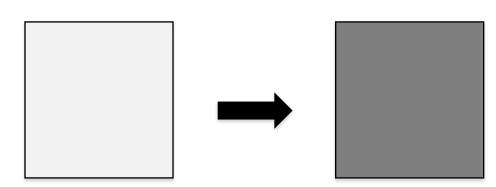


## **Corrosion Quantified**

- Corrosion is measured by sample weight gain
- Oxidation of alloys is measured experimentally using the <u>weight gain</u> in mg/dm<sup>2</sup>
- Oxide forms by incorporating oxygen in the metal structure causing an increase in weight
- 14.7 comes from density of O in ZrO2

$$W = (m_2 - m_1)/S$$

Dividing by the surface enables comparison between samples



Original coupon: Weight  $m_1$  Surface S

corroded coupon: Weight  $m_2 > m_1$ Surface S

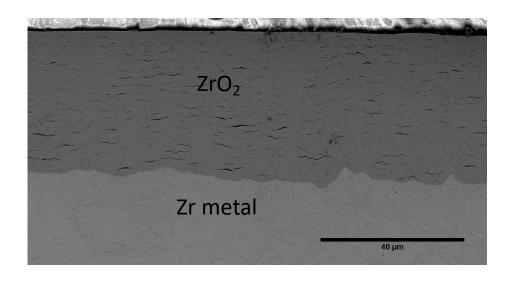
<u>Approximation weight gain – oxide thickness for Zr:</u>

$$\delta(\mu m) = \frac{w(mg/dm^2)}{14.7}$$

# Pilling-Bedworth ratio: PBR

- The ratio of the volume per unit of the metal oxide to the volume per unit of the corresponding metal is called the Pilling-Bedworth ratio (PBR)
- PBR < 1: the oxide coating layer is thin, likely broken and provides no protective effect (for example magnesium)
- PBR > 2: the oxide coating chips off and provides no protective effect (example iron)
- 1 < PBR < 2: the oxide coating is passivating and provides a protecting effect against further surface oxidation
- Passivation refers to a material becoming "passive," that is, less affected or corroded by the environment

$$m R_{PB} = rac{V_{oxide}}{V_{metal}} = rac{M_{oxide} \cdot 
ho_{metal}}{n \cdot M_{metal} \cdot 
ho_{oxide}}$$



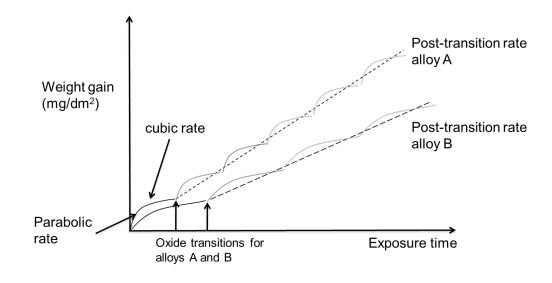
## **Example**

- A corrosion coupon of ZIRLO measures 2.8 cm x 2.8 cm x 600 µm and has an initial mass of 3 g. After corrosion for 200 days, its final mass is 3.0721 g. What is the estimated oxide thickness?
- First, we need to convert the units (10 cm = 1 dm)
- Find area
  - $S = 0.28*0.28 = 0.0784 dm^2$
- Find mass change
  - dM = 307.21 300 mg = 7.21 mg
- Then we calculate the weight gain per area
  - $-W = dM/S = (7.21)/0.0784 = 91.96 \text{ mg/dm}^2$
- Last estimate the thickness
  - d = W/14.7 = 91.96/14.7 = 6.3 microns thick after 200 days

## **Corrosion Rate**

- Average weight gain of a sample follows linear kinetics
- The oxide reaches transition at different times in different points, but the average is linear
- Critical oxide thickness for transition is defined as  $\delta^{\star} (\mu \text{m}) = 5.1 \exp \frac{-550}{T}$
- Critical time for transition is defined as

$$t^*(d) = 6.62 \times 10^{-7} \exp \frac{11949}{T}$$



After transition, oxide thickness is

$$\delta (\mu \mathbf{m}) = \delta^* + K_L (t - t^*)$$

$$K_L\left(\frac{\mu \mathrm{m}}{\mathrm{d}}\right) = 7.48 \times 10^6 \exp\frac{-12500}{T}$$

## **Example**

- Estimate the oxide thickness on a ZIRLO sample at 650 K after 200 days exposed to water.
- First, we have to determine if the oxide has gone through transition

$$t^* (d) = 6.62 \times 10^{-7} \exp \frac{11949}{T}$$

- $t^* = 6.62e-7*exp(11949/650 k) = 63.76 days$
- So, the sample is past transition, so we need to use the linear fit
- Next, we calculate the oxide thickness at transition  $\delta^{\star} (\mu m) = 5.1 \exp \frac{-550}{T}$

$$-\delta^{\star} (\mu m) = 5.1 \exp \frac{-550}{T}$$

- $-\delta^* = 5.1 \exp(-550/650) = 2.19$  microns
- Now, we can compute the final oxide thickness

$$K_L\left(\frac{\mu m}{d}\right) = 7.48 \times 10^6 \exp{\frac{-12500}{T}}$$

$$- K_L = 7.48e6*exp(-12500/650) = 0.0333$$

$$- \delta (\mu \mathbf{m}) = \delta^* + K_L (t - t^*)$$

$$-\delta = 2.19 + 0.0333*(200 - 63.76) = 6.73$$
 microns