

1. $T = 625 \text{ K}$ $t = 400 \text{ d}$ $t = \frac{500 \text{ mm}}{500 \times 10^{-4} \text{ cm}} = 0.5 \text{ cm}$

18/8 a.) $\delta = ?$ $\delta (\mu\text{m}) = \delta^* + K_L(t - t^*)$

$(\mu\text{m}) \delta^* = 5.1 \exp^{-\frac{550}{T}} = 5.1 \exp^{-\frac{550}{625}} = 2.115 \mu\text{m}$

$K_L (\mu\text{m}) = 7.48 \times 10^6 \exp^{-\frac{12500}{T}} = 7.48 \times 10^6 \exp^{-\frac{12500}{625}} = 0.0154 \mu\text{m}$

$t^* (\text{d}) = 6.62 \times 10^{-7} \exp^{\frac{11949}{T}} = 6.62 \times 10^{-7} \exp^{\frac{11949}{625}} = 133.007 \text{ d}$

$\delta = \delta^* + K_L(t - t^*) = 2.115 \mu\text{m} + 0.0154 \mu\text{m} (400 \text{ d} - 133 \text{ d})$
 $= 2.115 \mu\text{m} + 4.116 \mu\text{m}$

$\delta = 6.231 \mu\text{m}$ → that's like a lot right? actually it's thick

b.) $C_H^{\text{clad}} = \frac{2 F \delta \rho_{\text{oxide}} f_{\text{ZrO}_2}^0 \cdot M_H / M_O \times 10^6}{[t - \frac{\delta}{\text{PBR}}] \rho_{\text{metal}}}$

$\text{PBR} = 1.56$ $\rho_{\text{Zr}} = 6.5 \text{ g/cc}$ $\rho_{\text{ZrO}_2} = 5.68 \text{ g/cc}$

$f_{\text{ZrO}_2}^0 = \frac{32}{32 + 91} = 0.26$ $f = 18\%$

$t = 0.5 \text{ cm}$

$C_H^{\text{clad}} = \frac{2 (0.18 \times 5.68 \text{ g/cm}^3 \times 6.231 \times 10^{-4} \text{ cm} \times 0.26 \times 10^6 \text{ g/mol})}{[0.5 \text{ cm} - \frac{6.231 \times 10^{-4} \text{ cm}}{1.56}] (6.5 \text{ g/cm}^3)} \times 10^6$

$\delta = 6.231 \times 10^{-4} \text{ cm}$ $t = 0.5 \text{ cm}$

$C_H^{\text{clad}} = \frac{2 (0.18 \times 5.68 \text{ g/cm}^3 \times 6.231 \times 10^{-4} \text{ cm} \times 5.68 \text{ g/cm}^3 \times 0.26 \times 10^6 \text{ g/mol})}{[0.5 \text{ cm} - \frac{6.231 \times 10^{-4} \text{ cm}}{1.56}] (6.5 \text{ g/cm}^3)} \times 10^6$

$\delta = \delta^* + K_L(365 - 133)$

$\delta = 5.692 \mu\text{m}$
 $= 5.692 \times 10^{-4} \text{ cm}$

$C_H^{\text{clad}} = \frac{2.07 \times 10^{-5}}{3.2474} \times 10^6 = 6.376 \text{ wt ppm}$

$C_H^{\text{clad}} = \frac{1.891 \times 10^{-5}}{3.24976} \times 10^6$

$C_H^{\text{clad}} = 5.82 \text{ wt. ppm}$

- off by factor of 10
 - don't know where...

Alexandra Doherty

2. Diffusion of oxygen - through the oxide layer

3. PBR is the ratio of $\frac{M_{\text{oxide}} \rho_{\text{metal}}}{n \cdot M_{\text{metal}} \rho_{\text{oxide}}}$ and tells the thickness of the oxide layer relative to the metal Zr

$< 1 \rightarrow$ thin uniform oxide layer, minimal protection, likely broken

$1 < \text{PBR} < 2 \rightarrow$ passivating oxide layer

$> 2 \rightarrow$ thick and oxide may chip off, not much protection

4. Hydrides form within the oxide layer on the cladding due to the radiolysis or oxidation reaction with water $\text{Zr} + 2\text{H}_2\text{O} \rightarrow \text{ZrO}_2 + 2\text{H}_2$. They form 2x as much as oxides do (more fraction) hydrides can be radial / circumferential H is made, and migrate towards high stress \rightarrow which leads to APPT at the crack tip. Hydrogen is more brittle than ductile Zr, and causes hydride embrittlement. H embrittlement decreases ductility of the fuel clad and leads to further crack propagation

- twice the H is made, but H picks up \rightarrow like $\frac{1}{4}$

\rightarrow hydrides...

- hydride rim, slow diffusion

5. RIA = Reactivity Initiated Accidents \rightarrow large / rapid insertion of reactivity caused by ejection (PWR) or drop (BWR) of a control rod. This causes a rapid decrease in M adsorption of non-fuel and leads to a rapid power spike / rise in Temp. A fuel rod failure can also include radioactive material being released into coolant, causing rapid steam generation and pressure pulses which can damage the core / break internal pressure boundary

PWR \rightarrow ejection of control rod BWR \rightarrow drop control rod

- needed more of what happens to the material

Alexandra Donerty

6. LOCA \rightarrow loss of coolant Accident \rightarrow Different from RIA because it has a longer time scale (minutes) opposed to Severe RIA (0-1s) So there is a better chance at recovery.

LOCAs occur when a pipe bursts or something else occurs obstructing coolant flow. ^{lowering coolant} This leads to a $T \uparrow$ which increases ^{and ~~small~~ therm expansion} FG/FP release rate, increasing pressure. \uparrow Temp also increases oxidation by increasing diffusion rate and because oxidation is exothermic, it begins to "self-oxidize" causing further O/H embrittlement. More cracks, and therefore more metal exposed for oxidation. ~~Pressure increase period~~ As $T \uparrow$, α phase of Zr transitions to more brittle β phase. with a decrease in ductility, increase in pressure, and increase in cracking, cladding balloons and eventually bursts, releasing fuel outside the cladding. Can be ~~preferred~~ mitigated with ECCS (which can quench rod and lead to fragmentation of oxide) limits in place to control allowed amount of oxidation (177.)

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7. ductility is ^{already} reduced from H embrittlement, also experiences more corrosion/irradiation hardening. ^{4/5} Although in a LOCA, could go to a higher temp before burst and retain ductility for longer due to ^{less} slower change in T . ^{during LOCA} However T_m of fuel decreases in HBS (still $< 2750^\circ\text{C}$? I think?) So it is important to incorporate a changing T_m into the design so there is no meltdown
- most of it

Alexandra Donnelly

8. Improved reaction kinetics with steam \rightarrow ductile Zr based clad, also FeCrAl for passivation layer from steam oxidation

8/8 Improved cladding properties \rightarrow FeCrAl clad for formation of Al_2O_3 / ~~example~~ chromium oxide passivation layer

Improved fuel properties \rightarrow additives to \uparrow thermal conductivity

Enhanced fission product retention \rightarrow SiC \rightarrow temp resistant

9. High \uparrow steam leads to \uparrow oxidation rate \rightarrow exothermic and oxidizes parabolically until oxide layer is thick and loses passivation (PBR > 2) then linear / localized oxidation continues and breakaways can occur in oxides due to high strain

Paired with thick layer

- phase transitions

10. PCMI \rightarrow when ~~gap~~ gap closes and cladding hoop stress increases when reactivity is high. Determine gap param. can model FG pressure

4/6 Cladding oxidation / H pickup \rightarrow H gen $< 1\%$ possible throughout clad
Oxide limit 17% to prevent further embrittlement

Normal operating limits \rightarrow constraints on axial LHR to guarantee conditions don't exceed worst case

ALEXANDRA DONERTY

→ Chalk River Unidentified Deposit

11. CRUD is corrosion products composed of dissolved ions or solid particles (Ni, Fe, Co) which degrade heat production because they are slowly eroded by circulation of hot pressurized water. Then it can accelerate corrosion of cladding. ^{4/6} ^{fuel} halts performance and limits cladding lifespan. Heavy CRUD buildup on fuel clad surfaces from corrosion product transport - halfway

12. coordinated Boron and Lithium → Boric Acid is used to control reactivity, LiOH is used conversely to balance the pH so it is > 6.9, which is important because Ni/Fe solubility is pH dependent ^{4/6}

Zn injection is used to reduce radiation fields/inhibit stress corrosion cracking. (BWR) ~~Zn_{aq} promotes a protective spinel structured corrosion film on SS~~ (Due to Zn_{aq} promoting a protective spinel structured corrosion film on SS clad. ^{4/6} Corrosion products deposited on fuel, are activated and released into coolant) ← ignore if you want, I could be overexplaining