

Grain Boundary Energy and Surface Energy of α -U

Khadija Mahbuba,* Benjamin Beeler,*[†] and Andrea Jokissari,[†]

*North Carolina State University, Raleigh, NC 27607

[†]Idaho National Laboratory, Idaho Falls, ID 83415

INTRODUCTION

Metallic alloys of U-Zr and U-Pu-Zr are of interest for the Versatile Test Reactor and several microreactor designs. The α -U phase (stable up to 935 K [1]) exists in metallic fuel and has a significant contribution in fuel performance, however many fundamental material properties are still unknown. Behaviour study of grain boundary (GB) is one of them. GB has effect on plasticity [2], swelling [3], etc. of polycrystalline α -U. Symmetric tilt grain boundary (STGB) is a common type of GB, which has a mirror symmetry across the tilt plane (plane contains tilt axis). The energy cost to make a cavity along a STGB can be expressed as the energy difference of a STGB versus two surfaces of the same orientation [4]. Thus, the surface energy is also required to evaluate the mechanical performance of α -U. To analyze the interfacial energy (GB energy, surface energy, etc.) molecular dynamics along with a suitable interatomic potential, is a powerful tool [5]. In view of this, the current work has been conducted to evaluate the GB and surface energy of α -U at four different temperatures (300 K, 400 K, 500 K and 600 K) employing molecular dynamics to determine the variation of inter-facial energy as a function of STGB orientation and temperature.

METHODOLOGY

LAMMPS [6] along with UMo ADP potential [7] is utilized to perform molecular dynamics simulation. To generate the super-cell, simulation box is divided into two equal region, and region one is tilted with respect to defined tilt axis and tilt angle (tilt angle is half of misorientation angle). Taking the reflection of the former region across the tilt plane, region two is created. Hence a bicrystal of α -U with two STGBs is formed. In order to maintain the periodic boundary condition in all directions, supercell size varies with tilt axis, misorientation angle and shear plane. While investigating the surface energy, only region one with α -U atoms is created and region two is kept vacuum, thus two surfaces are created within a super-cell. An as-constructed STGB is shown in Figure 1.

A total of eighty STGBs are constructed, among them, twenty-four has tilt axis $\langle 1\ 0\ 0 \rangle$ (shear plane $(0\ 0\ 1)$), twenty-four has tilt axis $\langle 0\ 0\ 1 \rangle$ (shear plane $(0\ 1\ 0)$), and thirty-two has tilt axis $\langle 0\ 0\ 1 \rangle$ (shear plane $(1\ 0\ 0)$), such that entire scope of STGBs are studied with respect to three shear planes. These three categories are named as type A, type B, and type C respectively.

A Nose-Hoover barostat and Langevin thermostat is considered to relax the system, and after reaching equilibrium, observable parameters are collected. Systems are equilibrated for 200 ps and energies are averaged over final 50 ps. The inter-facial energies are calculated as-

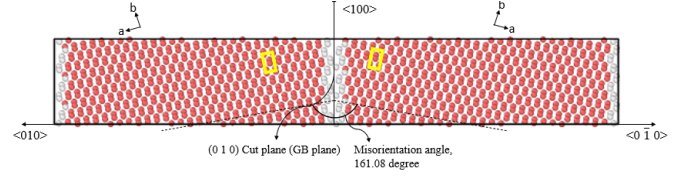


Fig. 1. The as-constructed α -U symmetric-tilt GB (STGB) ($\bar{1}\ 12\ 0$) with tilt axis $\langle 1\ 0\ 0 \rangle$ and shear plane $(0\ 0\ 1)$. Grain boundaries exist in the middle and on the edges of the super cell. Both red and white atoms are U, but with different coordination environments to illustrate grain boundaries. Yellow rectangles indicate the α -U unit cell and a and b represent the original $\langle 1\ 0\ 0 \rangle$ and $\langle 0\ 1\ 0 \rangle$ crystal structure orientation respectively.

are calculated as

$$E_{\text{inf}} = \frac{E^* - E_0}{A} \times N \quad (1)$$

where E_{inf} is the interfacial (STGB or surface) energy per unit interface (STGB or surface) area, E^* is the energy per atom of a system with an interface, N is the number of atoms in the system with an interface, E_0 is the energy per atom of a defect-free system, and A is the total area of the interface.

GB and surface energies are observed at 300 K, 400 K, 500 K, and 600 K where uranium is present in α phase.

RESULTS

GB energy variation with misorientation angles from 0° to 360° at 500 K temperature for type A, B, and C STGBs are presented in Figure 2. Data points are connected via straight line to guide the eye. Because of the mirror symmetry of α -U along $(1\ 0\ 0)$ and $(0\ 0\ 1)$ planes [8], no GB is formed at 180° misorientation angle of type A, B, and C STGBs and 0° misorientation angle of only type B STGBs.

Number of cusps over the GB energy landscape is subjective to type of STGBs. For instance, type A STGBs ($\langle 1\ 0\ 0 \rangle$ tilt axis) have three deep cusps at the $(0\ 1\ 0)$, $(\bar{3}\ \bar{12}\ 0)$, and $(0\ \bar{1}\ 0)$ plane and also a minor cusp at plane $(\bar{3}\ 6\ 0)$. The GB energy profile of type B STGBs ($\langle 0\ 0\ 1 \rangle$ tilt axis) only has a minor cusp at the $(\bar{2}\ 0\ \bar{1})$ plane. However, type C STGBs ($\langle 0\ 0\ 1 \rangle$ tilt axis) have the highest number of cusps (eight), which are oriented at a mirror symmetry orientation with respect to the $\langle 0\ 0\ 1 \rangle$ tilt axis, see Figure 2. Major cusps of type C STGBs are also mirror oriented oriented with respect to $(0\ 0\ 1)$ STGB plane, between these two cusps $(0\ 18\ \bar{25})$ STGB plane has lowest GB energy ($0.36\ \text{J/m}^2$) and $(0\ \bar{18}\ 25)$ plane possesses highest GB energy ($1.39\ \text{J/m}^2$) among type C STGBs.

This study indicates that type C STGBs (0.94 J/m^2) have the highest average GB energy, followed by type A STGBs (0.78 J/m^2), with type B STGBs (0.75 J/m^2) exhibiting the lowest GB energies. Among the entire dataset of eighty STGBs, the lowest GB energy is observed for the type A STGB at the

$(\bar{3} \ 12 \ 0)$ STGB plane, where $\langle 1 \ 0 \ 0 \rangle$ is the tilt axis, $(0 \ 0 \ 1)$ is the shear plane, and the GB energy is 0.30 J/m^2 . Low energy STGBs would be a likely candidate to form twin. The highest

GB energy is found for a type C STGB at the $(0 \ \bar{18} \ 25)$ STGB plane, where $\langle 0 \ 0 \ 1 \rangle$ is the tilt axis, $(1 \ 0 \ 0)$ is the shear plane, and the GB energy is 1.39 J/m^2 .

Effect of temperature on GB energy of α -U STGBs is studied considering twenty-two STGBs (eight from type A, seven from type B, and seven from type C) and presented in Figure 3. All three types (A, B, and C) STGBs experience an increase in GB with temperature (specially from 400 K to 600 K). However average change in GB energy within the studied temperature range (300 K to 600 K) with respect to GB energy attained at 300 K varies with STGB type; for example, type A by 15.8%, type B by 7.8%, and type C by 82%.

In the current work, surfaces which are only related to selected STGBs of α -U are studied. Considering three primary surfaces, surface energy of total eighty-three surfaces are calculated. Surfaces are classified as per the classification of STGBs. From Figure 4, the trend of surface energy of type A, B, and C surfaces with tilt angle is observed at 500 K.

No discernible trend is found for type A and C surfaces whereas type B surface has a mirror symmetry with respect to $(1 \ 0 \ 0)$ plane. Similar pattern in GB energy plot could be attained if minor cusp is omitted, see Figure 2. Type A, B and C surfaces have an average surface energy 1.25 J/m^2 , 1.21 J/m^2 , and 1.26 J/m^2 respectively. Among the studied surface, the maximum surface energy has been observed at the type C

$(0 \ \bar{18} \ 25)$ plane with a surface energy of 1.52 J/m^2 and lowest energy for the type A $(\bar{3} \ 12 \ 0)$ plane with a surface energy of 1.0 J/m^2 . At 500 K, surface energy of three primary surfaces ($(1 \ 0 \ 0)$, $(0 \ 1 \ 0)$, and $(0 \ 0 \ 1)$) are calculated as 1.13 J/m^2 , 1.39 J/m^2 , and 1.11 J/m^2 respectively.

An investigation of surface energy variation with temperature is performed for a subset of STGBs, similar to the investigation of GB energy with temperature and plotted in Figure 5. Like GB energy, a general positive correlation between surface energy and temperature (specifically from 400 K to 600 K) is observed. Comparing among three types of surfaces, type C is significantly more sensitive to temperature as average change is 9.7%, 2.5%, and 4.4% with respect to surface energy attained at 300 K for type C, B, and A surfaces respectively.

From Figures 2 and 4, it is also observed that GB energy of anisotropic α -U is not only dependent on misorientation (or tilt) angle but also on tilt axis and shear plane because of the unique crystallography of α -U. Comparing the average interfacial energy of STGBs of α -U with other U-based system at 600 K (for the misorientation domain 0° to 90°), γ -U has lowest GB energy, followed by α -U and U_3Si_2 [9, 10]. Similar result is found for surface energy.

CONCLUSIONS

Elucidating the dependence of crystallographic orientation and temperature on GB and surface energy, the current work is the most comprehensive study on inter-facial energy of α -U to authors knowledge. At 500 K, the global arithmetic average of the surface energy (1.24 J/m^2) is approximately 1.5 times the GB energy (0.82 J/m^2). With the increase of temperature, GB energy in general increases more rapidly than the surface energy. The information from this work could be utilized to study GB evolution in polycrystalline α -U and deformation under irradiation, leading to the development of improved fuel performance models.

ACKNOWLEDGMENTS

Work supported through the INL Laboratory Directed Research and Development (LDRD) Program under DOE Idaho Operations Office Contract DE-AC07-05ID14517.

REFERENCES

1. G. L. HOFMAN, *Metallic Fuels*, American Cancer Society, pp. 1–53 (2015).
2. A. JOKISAARI, "Irradiation-induced internal stresses in polycrystalline α -uranium: a mesoscale mechanical approach," *Computational Materials Science*, **176**, 109545 (2020).
3. J. REST, "Kinetics of fission-gas-bubble-nucleated void swelling of the alpha-uranium phase of irradiated U-Zr and U-Pu-Zr fuel," *Journal of Nuclear Materials*, **207**, 192 – 204 (1993).
4. E. BOURASSEAU, A. MOURET, P. FANTOU, X. ILLIS, and R. BELIN, "Experimental and simulation study of grain boundaries in UO_2 ," *Journal of Nuclear Materials*, **517** (02 2019).
5. G. S. ROHRER, "Grain boundary energy anisotropy: a review," *Journal of Materials Science*, **46**, 5881 – 5895 (2011).
6. S. PLIMPTON, "Fast Parallel Algorithms for Short-Range Molecular Dynamics," *J. Comp. Phys.*, **117**, 1–19 (1995).
7. S. STARIKOV, L. KOLOTOVA, A. KUKSIN, D. SMIRNOVA, and V. TSEPLYAEV, "Atomistic simulation of cubic and tetragonal phases of U-Mo alloy: Structure and thermodynamic properties," *Journal of Nuclear Materials*, **499**, 451 – 463 (2018).
8. E. Y. CHEN, C. DEO, and R. DINGREVILLE, "Atomistic simulations of temperature and direction dependent threshold displacement energies in α - and γ -uranium," *Computational Materials Science*, **157**, 75 – 86 (2019).
9. B. BEELER, Y. ZHANG, and Y. GAO, "An atomistic study of grain boundaries and surfaces in γ -U-Mo," *Journal of Nuclear Materials*, **507**, 248 – 257 (2018).
10. B. BEELER, M. BASKES, D. ANDERSSON, M. W. COOPER, and Y. ZHANG, "Molecular dynamics investigation of grain boundaries and surfaces in U_3Si_2 ," *Journal of Nuclear Materials*, **514**, 290 – 298 (2019).

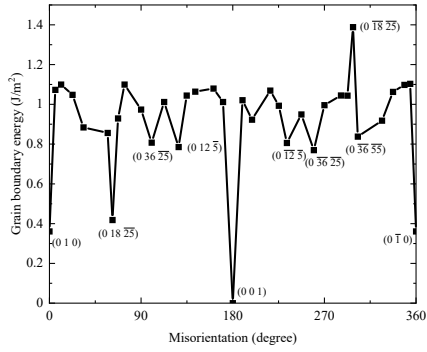
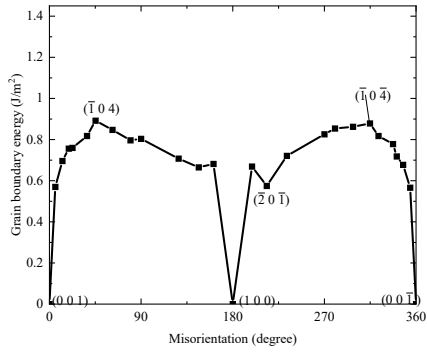
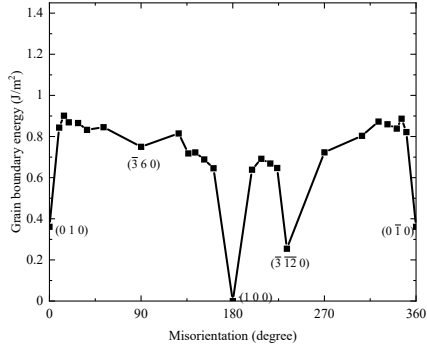


Fig. 2. GB energy of STGB (symmetric tilt grain boundary) of (a) type A, (b) type B, and (c) type C as a function of misorientation angles at 500 K. Key features are labeled with their STGB plane orientation.

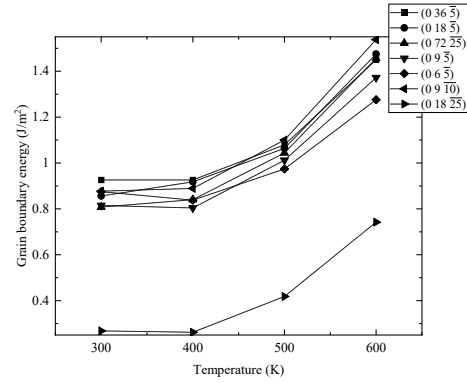
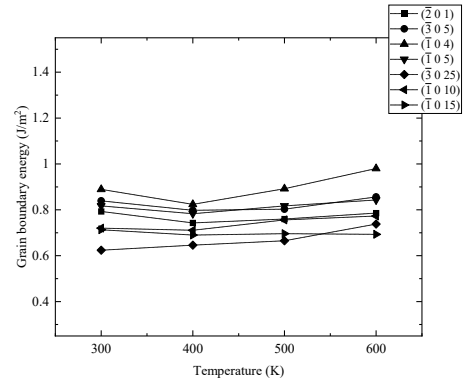
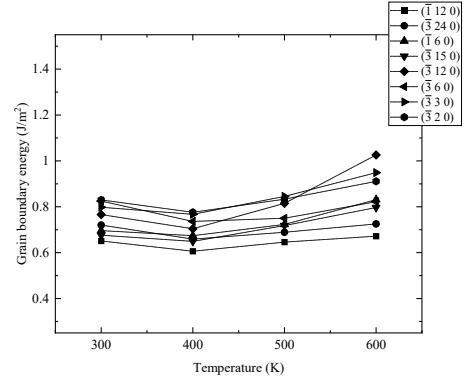


Fig. 3. GB energy of selected STGBs of (a) type A, (b) type B, and (c) type C as a function of temperature

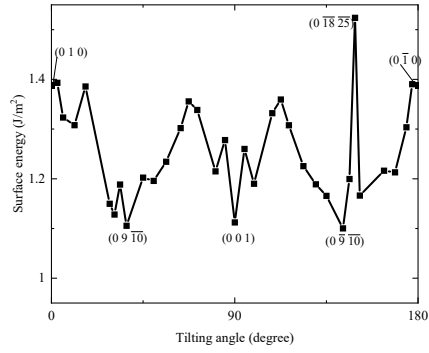
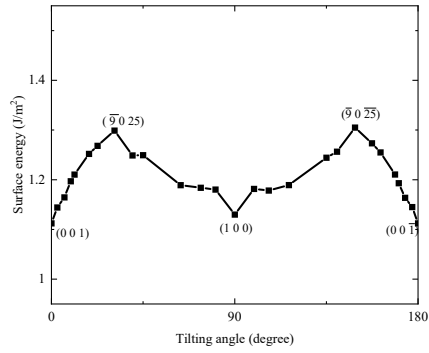
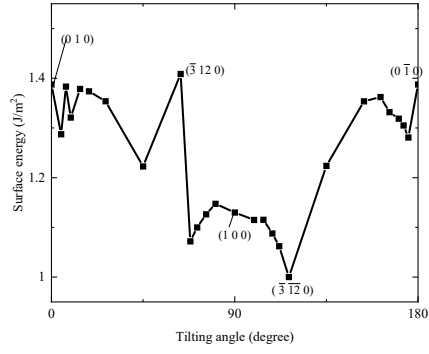


Fig. 4. Surface energy of surfaces of (a) type A, (b) type B, and (c) type C as a function of tilt angle at 500 K.

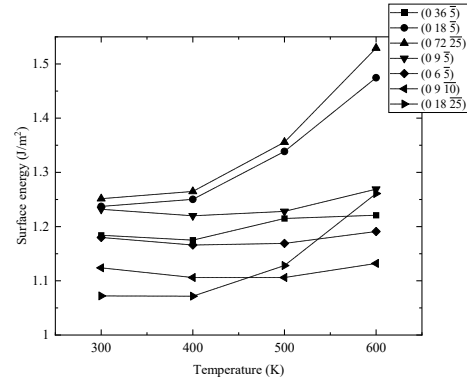
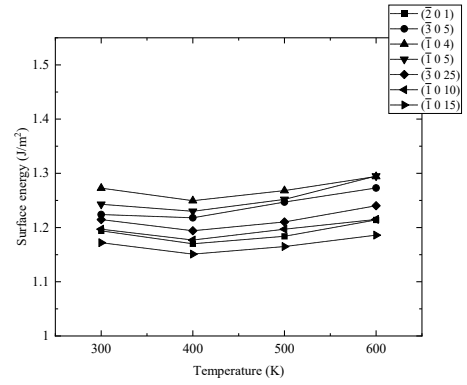
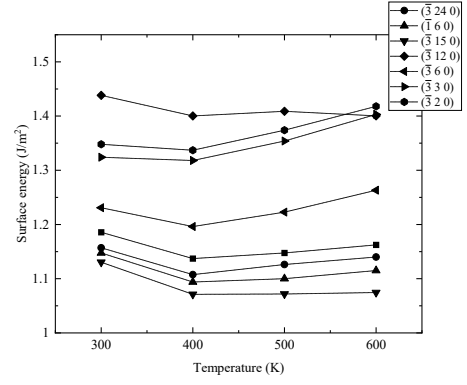


Fig. 5. Surface energy of selected surfaces of (a) type A, (b) type B, and (c) type C as a function of temperature.