

Hyperuniformity in 1D Quasiperiodic Point Patterns

Numerical Extraction of the Scaling Exponent
via Diffusion Spreadability

Presented to Professor Salvatore Torquato

Princeton University

February 25, 2026

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Hyperuniformity in 1D Quasicrystals

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This presentation covers four completed phases of the 1D hyperuniformity project: code validation, pattern generation, real-space variance analysis, and reciprocal-space spreadability analysis. The main result is the numerical confirmation that $\alpha = 3$ universally for all metallic-mean substitution tilings.

Outline

- 1 Background
- 2 Phase 1: Code Validation
- 3 Phase 2: Pattern Generation
- 4 Phase 3: Real-Space Analysis
- 5 Phase 4: Two-Phase Media & Spreadability
- 6 Technical Challenges
- 7 Summary & Future Work

2026-02-25

Hyperuniformity in 1D Quasicrystals

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1 Background
2 Phase 1: Code Validation
3 Phase 2: Pattern Generation
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6 Technical Challenges
7 Summary & Future Work

What Is Hyperuniformity?

A point pattern is **hyperuniform** if local number fluctuations are *suppressed* relative to random:

$$\sigma^2(R) \sim R^{d-\alpha}, \quad S(k) \sim |k|^\alpha \text{ as } k \rightarrow 0$$

Class	Exponent	Variance growth	Examples
I	$\alpha > 1$	R^{d-1} (surface area)	crystals, quasicrystals
II	$\alpha = 1$	$R^{d-1} \ln R$	period-doubling chains
III	$0 < \alpha < 1$	$R^{d-\alpha}$	certain aperiodic chains

Goal: numerically extract α for three 1D quasicrystal families and verify $\alpha = 3$.

Challenge: dense Bragg peaks make direct $S(k)$ measurement impossible \implies use *diffusion spreadability* (Torquato, 2021).

2026-02-25

Hyperuniformity in 1D Quasicrystals

- Background
- What Is Hyperuniformity?

Hyperuniformity was introduced by Torquato & Stillinger (2003). The key idea: for a Poisson process $\sigma^2 \sim R^d$ (volume scaling), while hyperuniform systems grow *slower*. The exponent α classifies how much slower. Class I ($\alpha > 1$) is the strongest form, with variance scaling like the surface area of the window.

The “measurement problem” for quasicrystals motivates the two-phase media / spreadability approach that is the heart of Phase 4.

What Is Hyperuniformity?

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The Three Metallic-Mean Quasicrystal Chains

Chain	Rule	Matrix M	Metallic mean μ	ρ
Fibonacci	$S \rightarrow L, L \rightarrow LS$	$\begin{pmatrix} 0 & 1 \\ 1 & 1 \end{pmatrix}$	$\tau \approx 1.618$	0.724
Silver	$S \rightarrow L, L \rightarrow LLS$	$\begin{pmatrix} 0 & 1 \\ 1 & 2 \end{pmatrix}$	$\mu_2 \approx 2.414$	0.500
Bronze	$S \rightarrow L, L \rightarrow LLLS$	$\begin{pmatrix} 0 & 1 \\ 1 & 3 \end{pmatrix}$	$\mu_3 \approx 3.303$	0.361

Theoretical prediction (eigenvalue formula):

$$\alpha = 1 - \frac{2 \ln |\lambda_2|}{\ln \lambda_1}$$

Since $\det(\mathbf{M}) = -1$ for all three $\implies |\lambda_2| = 1/\lambda_1 \implies \alpha = 3$ (universal)

Tile lengths: $S = 1, L = \mu$. Production chains: $N \sim 10^7$ tiles.

2026-02-25

Hyperuniformity in 1D Quasicrystals

└ Background

└ The Three Metallic-Mean Quasicrystal Chains

Each chain is defined by a 2×2 substitution matrix acting on tile counts $[n_S, n_L]$. The eigenvalues determine both the growth rate (largest eigenvalue $\lambda_1 = \mu$) and the hyperuniformity exponent via the second eigenvalue.

Because $\det(\mathbf{M}) = -1$ for all metallic-mean matrices, $|\lambda_2| = 1/\mu$, giving $\alpha = 1 + 2 = 3$ universally. This is a purely algebraic result — our job is to verify it numerically.

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Figure 1: Poisson Variance Benchmark

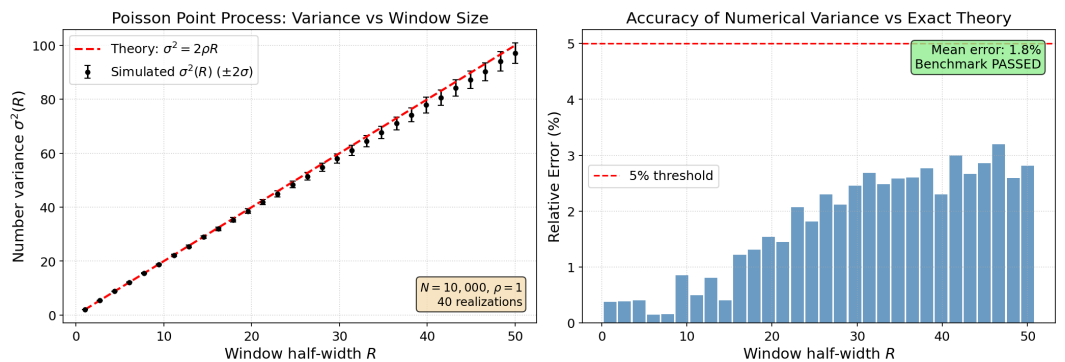
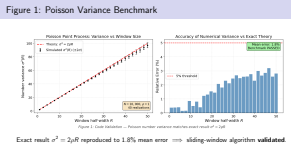


Figure 1: Code Validation — Poisson number variance matches exact result $\sigma^2 = 2\rho R$

Exact result $\sigma^2 = 2\rho R$ reproduced to 1.8% mean error \implies sliding-window algorithm **validated**.

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Hyperuniformity in 1D Quasicrystals
└ Phase 1: Code Validation

└ Figure 1: Poisson Variance Benchmark



What this shows: We generate 40 independent Poisson point patterns ($N=10,000$, $\rho=1$) and compute $\sigma^2(R)$ via binary-search sliding windows.

Left panel: Computed variance (black dots $\pm 2\sigma$) vs. exact theory $\sigma^2 = 2\rho R$ (red dashed).

Right panel: Relative error at each R , all below 5%. Green box shows mean error = 1.8%.

Why it matters: This same algorithm is used for all subsequent variance computations. Validating it against the Poisson exact result ensures we can trust the quasicrystal measurements.

Two Independent Generation Methods

Method 1: Substitution

- Iterate substitution rules from seed L
- 34 iterations \Rightarrow Fibonacci
 $N=14,930,352$
- Generates 10^7 tiles in ~ 1 s
- Exactly 2 distinct spacings (verified)

Method 2: Cut-and-Project

- Embed \mathbb{Z}^2 lattice
- Strip of width ω along slope $1/\mu$
- Project interior points onto 1D line
- $\omega = \mu$: Class I; $\omega \neq \mu$: Class II

Both methods produce the **same quasicrystal** (up to rescaling), allowing cross-validation of $\bar{\Lambda}$.

Chain	Density ρ	Spacing ratio L/S
Fibonacci	0.7236	$\tau \approx 1.618$
Silver	0.5000	$\mu_2 \approx 2.414$
Bronze	0.3613	$\mu_3 \approx 3.303$

2026-02-25

Hyperuniformity in 1D Quasicrystals

- └ Phase 2: Pattern Generation

Two Independent Generation Methods

The substitution method is conceptually simple: start with L , apply the rule, repeat. The cut-and-project method provides an independent realization at a different density, which is crucial for verifying the rescaling invariance of $\bar{\Lambda}$.
Key point: substitution gives $\rho \approx 0.72$ for Fibonacci while projection gives $\rho \approx 0.85$, yet both yield the same $\bar{\Lambda} = 0.200$.

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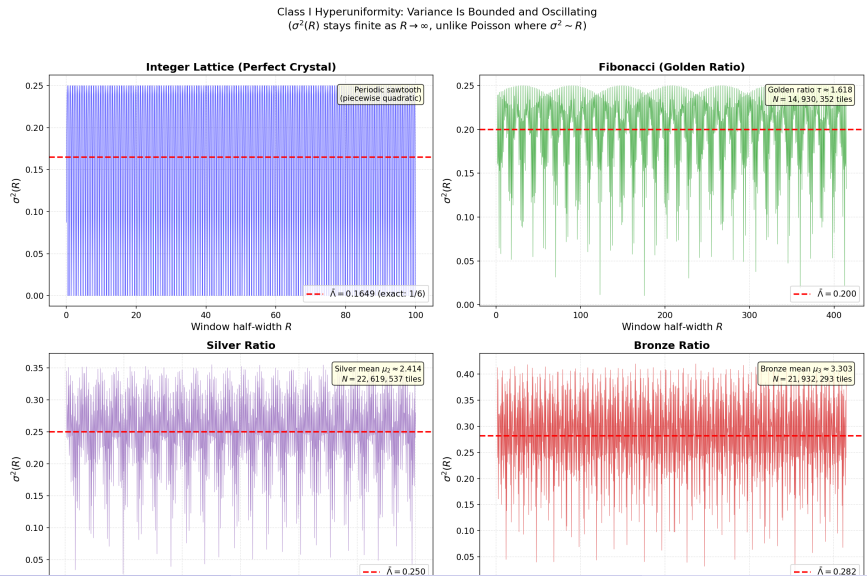
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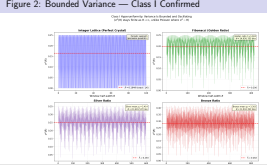
Figure 2: Bounded Variance — Class I Confirmed



2026-02-25

Hyperuniformity in 1D Quasicrystals
└ Phase 3: Real-Space Analysis

└ Figure 2: Bounded Variance — Class I Confirmed



What this shows: Number variance $\sigma^2(R)$ for the integer lattice (top-left) and three quasicrystal chains. Each curve: 30,000 random windows at 1,000 R -values.

Key observation: All four curves are *bounded and oscillating* — they do not grow with R . This is the defining signature of Class I hyperuniformity.

Red dashed lines: $\bar{\Lambda}$ (surface-area coefficient). Lattice: $\bar{\Lambda} \approx 1/6$. Values increase monotonically with μ : Fibonacci (0.200) < Silver (0.250) < Bronze (0.282).

Interpretation: More “aperiodic” chains (higher μ) have larger fluctuation amplitudes, but all remain Class I.

Surface-Area Coefficient $\bar{\Lambda}$

Pattern	N (tiles)	$\bar{\Lambda}$	vs Lattice
Integer Lattice	100,000	0.165 ($\approx 1/6$)	reference
Fibonacci (substitution)	14,930,352	0.200	1.21×
Silver (substitution)	22,619,537	0.250	1.51×
Bronze (substitution)	21,932,293	0.282	1.71×
Fibonacci (projection)	220,161	0.200	1.21×

- $\bar{\Lambda}$ increases monotonically with metallic mean μ
 - Substitution vs. projection agree to **0.1%** for Fibonacci
- ⇒ $\bar{\Lambda}$ is **density-independent** (rescaling invariance confirmed)

2026-02-25

Hyperuniformity in 1D Quasicrystals

- └ Phase 3: Real-Space Analysis
- └ Surface-Area Coefficient $\bar{\Lambda}$

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■ Substitution vs. projection agree to **0.1%** for Fibonacci
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The projection method generates Fibonacci at $\rho \approx 0.85$ while substitution gives $\rho \approx 0.72$. Despite the density difference, $\bar{\Lambda}$ agrees to 0.1%. Under rescaling $x \rightarrow ax$, we have $\sigma'^2(R) = \sigma^2(R/a)$ whose long-range average is unchanged. This numerical verification confirms the theoretical expectation.

Figure 3: Hyperuniformity Test — $\sigma^2(R)/R \rightarrow 0$

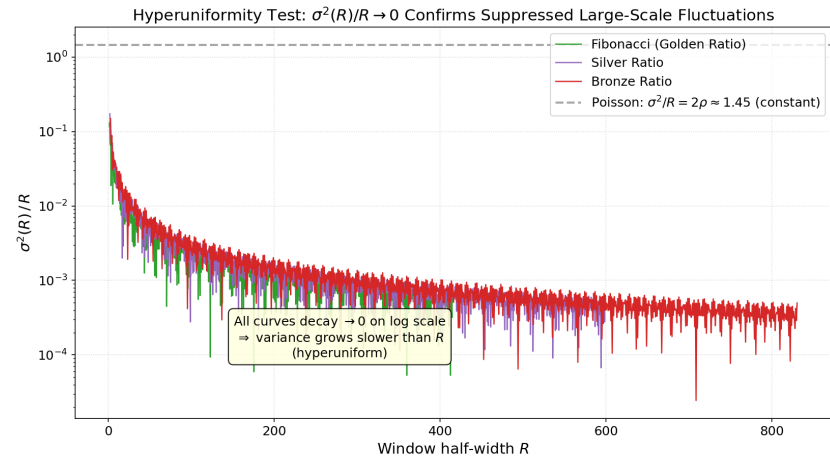
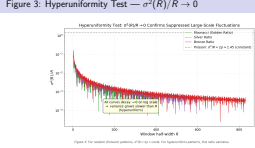


Figure 3: For random (Poisson) patterns, $\sigma^2/R = 2\rho = \text{const}$. For hyperuniform patterns, this ratio vanishes.

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Hyperuniformity in 1D Quasicrystals
└ Phase 3: Real-Space Analysis

└ Figure 3: Hyperuniformity Test — $\sigma^2(R)/R \rightarrow 0$



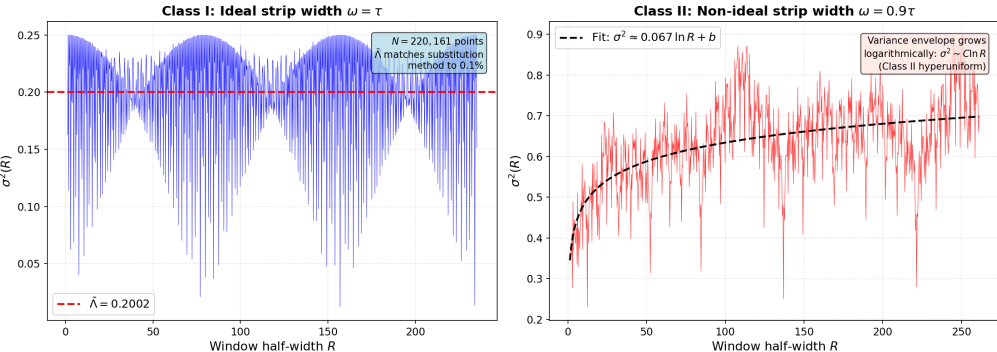
What this shows: The ratio $\sigma^2(R)/R$ on a log scale for all three chains. For Poisson, this ratio equals $2\rho \approx 1.45 = \text{const}$ (gray dashed line at top). For any hyperuniform pattern, it must decay to zero.

Key observation: On the log y -axis, the decay from $\sim 10^{-1}$ to $\sim 10^{-3}$ is clearly visible. All three chains show power-law-like decay, confirming that number fluctuations grow much slower than the window size.

Why it matters: This is the simplest direct real-space test of hyperuniformity. The three orders-of-magnitude gap between the Poisson constant and the quasicrystal curves at large R is the signature of suppressed fluctuations.

Figure 4: Projection — Class I vs. Class II

Projection (Cut-and-Project) Method: Strip Width Controls Hyperuniformity Class
Ideal $\omega = \tau$ gives Class I (bounded); non-ideal $\omega \neq \tau$ degrades to Class II (log growth)

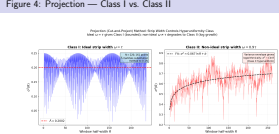


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Hyperuniformity in 1D Quasicrystals

Phase 3: Real-Space Analysis

Figure 4: Projection — Class I vs. Class II

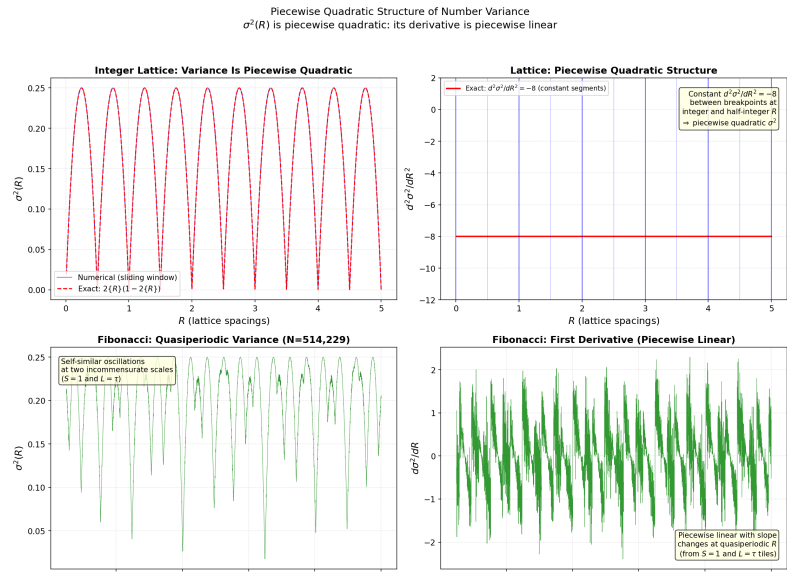


Left panel (Class I): Ideal strip width $\omega = \tau$. Variance is bounded with $\bar{\Lambda} = 0.200$, matching the substitution result to 0.1%.

Right panel (Class II): Non-ideal strip width $\omega = 0.9\tau$. The variance envelope *grows logarithmically*: $\sigma^2 \sim 0.067 \ln R + b$. This logarithmic growth is the hallmark of Class II hyperuniformity.

Takeaway: The hyperuniformity class is controlled by the projection strip width. Any deviation from the ideal $\omega = \mu$ degrades Class I to Class II.

Figure 5: Piecewise Quadratic Structure

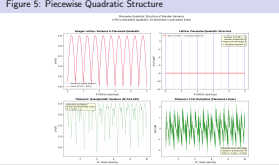


2026-02-25

Hyperuniformity in 1D Quasicrystals

└ Phase 3: Real-Space Analysis

└ Figure 5: Piecewise Quadratic Structure



Top-left: Lattice variance at high resolution, overlaid with exact formula $\sigma^2(R) = 2\{R\}(1 - 2\{R\})$. Perfect agreement.

Top-right: The exact second derivative $d^2\sigma^2/dR^2 = -8$ is constant between breakpoints at every half-integer R . Vertical lines mark these breakpoints. This confirms σ^2 is piecewise quadratic.

Bottom-left: Fibonacci variance shows self-similar oscillations at two incommensurate scales ($S=1$ and $L=\tau$).

Bottom-right: Fibonacci first derivative $d\sigma^2/dR$ is piecewise linear with slope changes at quasiperiodic positions determined by S and L tile lengths — confirming the piecewise quadratic structure with a dense set of breakpoints.

The Two-Phase Media Approach

Problem: $S(k)$ has dense Bragg peaks at irrational $k \implies$ cannot read off α directly.

Solution — four steps:

- 1 **Decorate** each point with a solid rod (half-length $a = \phi_2/2\rho$, $\phi_2 = 0.35$)
- 2 **Compute** spectral density: $\tilde{\chi}_V(k) = \rho \left(\frac{2\sin ka}{k}\right)^2 S(k)$
- 3 **Evaluate** excess spreadability: $E(t) = \frac{\Delta k}{\pi\phi_2} \sum_n \tilde{\chi}_V(k_n) e^{-k_n^2 Dt}$
- 4 **Extract** α from long-time decay: $\alpha(t) = -2 \frac{d\ln E}{d\ln t} - 1$

Chain	ρ	Rod $2a$	Min gap	Overlap?
Fibonacci	0.724	0.484	1.000	No
Silver	0.500	0.700	1.000	No
Bronze	0.361	0.969	1.000	No

2026-02-25

Hyperuniformity in 1D Quasicrystals

- Phase 4: Two-Phase Media & Spreadability
- The Two-Phase Media Approach

The two-phase media approach was introduced by Torquato (2021). The key insight: the Gaussian kernel $e^{-k^2 Dt}$ in the spreadability integral acts as a natural smoother of the Bragg peak spectrum. At time t , it samples wavevectors $k \lesssim 1/\sqrt{Dt}$. The power-law tail of the Bragg peak envelope then translates into a power-law decay of $E(t)$, from which α can be extracted. Non-overlap is verified: $2a < \min(\text{spacings}) = 1.0$ for all chains. The packing fraction $\phi_2 = 0.35$ is chosen following Hitin-Bialus et al. (2024).

The Two-Phase Media Approach

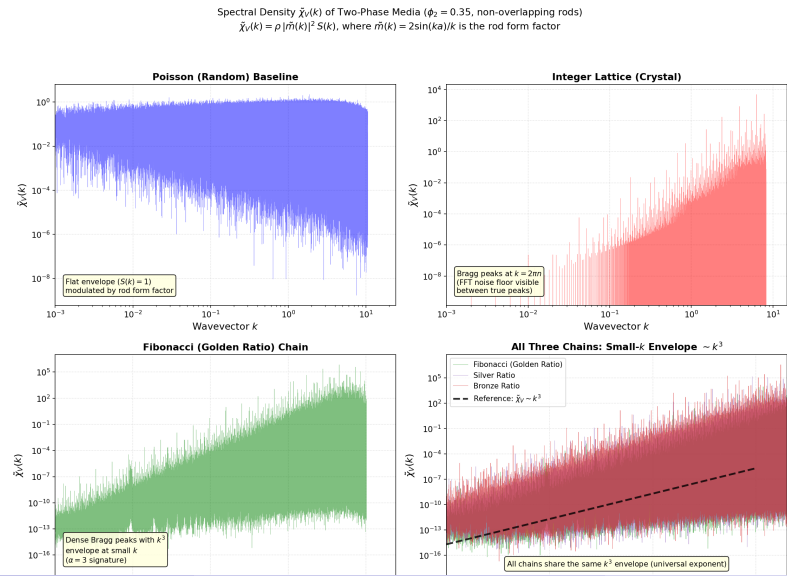
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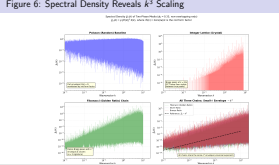
Figure 6: Spectral Density Reveals k^3 Scaling



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Hyperuniformity in 1D Quasicrystals

- Phase 4: Two-Phase Media & Spreadability
- Figure 6: Spectral Density Reveals k^3 Scaling



Top-left (Poisson): Flat $S(k) = 1$ envelope, modulated only by the rod form factor $|\tilde{m}(k)|^2$. No hyperuniformity.

Top-right (Lattice): Bragg peaks at $k = 2\pi n$. The FFT computation introduces a noise floor between the true peaks (this does not affect results since the lattice spreadability is computed analytically).

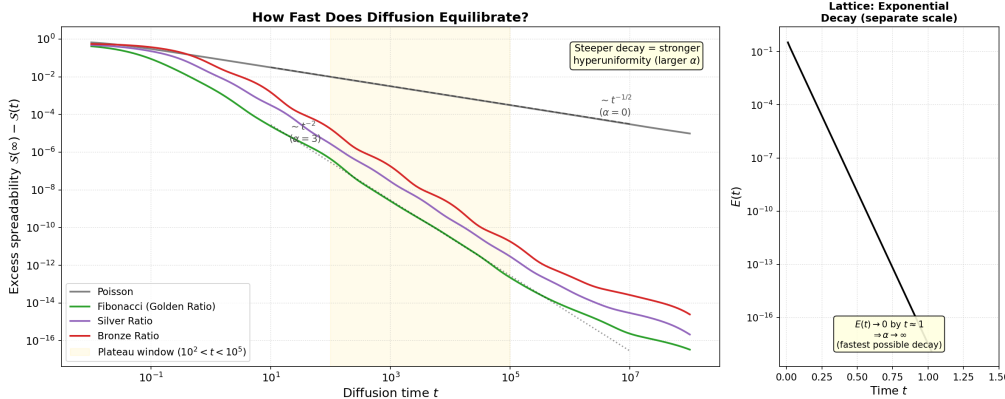
Bottom-left (Fibonacci): Dense Bragg peaks. The envelope of peak heights follows k^3 at small k — the spectral signature of $\alpha = 3$.

Bottom-right (All chains overlaid): All three chains share the same k^3 envelope (black dashed reference line), confirming universality.

The k^3 scaling is the Fourier-space manifestation of $\alpha = 3$ Class I hyperuniformity.

Figure 7: Excess Spreadability — Decay Rates

Excess Spreadability $S(\infty) - S(t)$: Decay rate reveals α — at long times $E(t) \sim t^{-(1+\alpha)/2}$

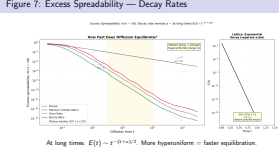


At long times: $E(t) \sim t^{-(1+\alpha)/2}$. More hyperuniform = faster equilibration.

2026-02-25

Hyperuniformity in 1D Quasicrystals
└ Phase 4: Two-Phase Media & Spreadability

└ Figure 7: Excess Spreadability — Decay Rates



Main panel (left): Log-log plot of excess spreadability $E(t)$ for Poisson and the three quasicrystals.

Gray (Poisson): Slow decay $\sim t^{-1/2}$ ($\alpha = 0$).

Green/Purple/Red (quasicrystals): Steeper decay $\sim t^{-2}$ ($\alpha = 3$). The gold shaded region marks the measurement window $[10^2, 10^5]$.

Right panel (Lattice): Plotted on separate axes because the lattice $E(t)$ decays *exponentially* (reaches zero by $t \approx 1$), confirming $\alpha \rightarrow \infty$. If plotted on the same axes, it would compress all other curves to a thin band.

Hierarchy: Random < quasicrystal < crystal in terms of equilibration speed.

Main Result: All Three Chains Converge to $\alpha = 3$

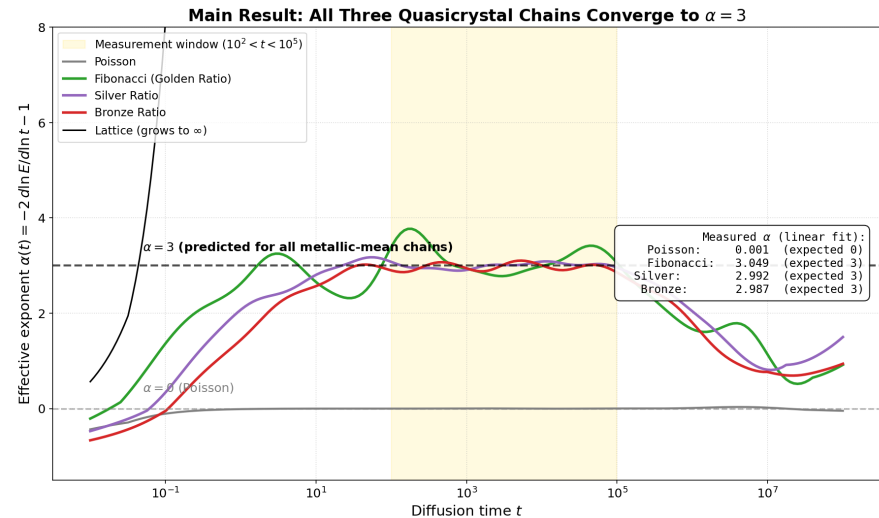
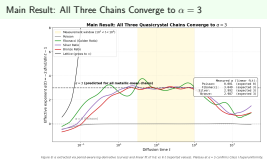


Figure 8: α extracted via period-aware log-derivative (curves) and linear fit of $\ln E$ vs $\ln t$ (reported values). Plateau at $\alpha = 3$ confirms Class I hyperuniformity.

2026-02-25

Hyperuniformity in 1D Quasicrystals
└ Phase 4: Two-Phase Media & Spreadability

└ Main Result: All Three Chains Converge to $\alpha = 3$



This is the central result of the project.

The effective exponent $\alpha(t) = -2 d \ln E / d \ln t - 1$ is plotted versus diffusion time t . Smoothing uses a period-aware sliding window matched to each chain's oscillation period $2 \ln \mu$ in $\ln t$ space (cancels oscillatory contributions).

Gray (Poisson): Flat at $\alpha \approx 0$ — correct.

Black (Lattice): Grows without bound — $\alpha \rightarrow \infty$.

Green/Purple/Red: All plateau at $\alpha \approx 3$ within the gold-shaded measurement window $[10^2, 10^5]$.

Measured values (linear fit of $\ln E$ vs $\ln t$): Poisson = 0.001, Fibonacci = 3.049, Silver = 2.992, Bronze = 2.987.

All three metallic-mean chains confirm $\alpha = 3$ via an independent dynamical measurement, matching the eigenvalue prediction.

Extracted α Values

Pattern	α (measured)	Expected	Status
Poisson	0.001	0	✓
Integer Lattice	exp. decay	∞	✓
Fibonacci (Golden Ratio)	3.049	3	✓
Silver Ratio	2.992	3	✓
Bronze Ratio	2.987	3	✓

All three metallic-mean chains yield $\alpha \approx 3$, confirming the universal eigenvalue prediction via an independent dynamical measurement (diffusion spreadability).

2026-02-25

Hyperuniformity in 1D Quasicrystals

└ Phase 4: Two-Phase Media & Spreadability

└ Extracted α Values

Values extracted via linear fit of $\ln E$ vs $\ln t$ over $[10^2, 10^5]$ (more robust than pointwise median, averages over oscillations). Errors: Fibonacci 1.6%, Silver 0.3%, Bronze 0.4%. These are consistent with finite-size effects from the FFT grid.

The two benchmarks (Poisson $\alpha = 0$ and lattice $\alpha \rightarrow \infty$) bracket the quasicrystal results and confirm the method is working correctly at both extremes.

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All three metallic-mean chains yield $\alpha \approx 3$, confirming the universal eigenvalue prediction via an independent dynamical measurement (diffusion spreadability).

1. System size sensitivity (non-monotonic convergence):

- FFT grid $k_n = 2\pi n/L$ misaligns with irrational Bragg peaks
- Fibonacci ($\rho=0.72$, smallest L for given N) needs $N \sim 10^7$
- Convergence is *non-monotonic*: $N=500k$ gives $\alpha=1.5$, $N=5.7M$ gives $\alpha=1.5$, $N=14.9M$ gives $\alpha=3.0$

2. Integer lattice artifact:

- FFT histogram binning creates noise floor mimicking $k^3 \Rightarrow$ spurious $\alpha \approx 3$
- **Fix:** analytical Bragg peak formula $E(t) = \frac{2\rho}{\phi_2} \sum_{n=1}^{50} |\tilde{m}(2\pi n)|^2 e^{-(2\pi n)^2 Dt}$

3. Noisy log derivative:

- Point-by-point `np.gradient` amplifies Bragg peak noise
- **Fix 1:** Period-aware sliding window matched to oscillation period $2 \ln \mu$ (cancels oscillatory contributions)
- **Fix 2:** Linear fit of $\ln E$ vs $\ln t$ over plateau window for single robust α value

2026-02-25

Hyperuniformity in 1D Quasicrystals
└ Technical Challenges

└ Technical Challenges & Solutions

These challenges consumed the majority of the Phase 4 development time. The non-monotonic convergence was the most surprising finding: certain system sizes give *worse* results than smaller ones because of how the FFT grid happens to align (or misalign) with the dominant Bragg peaks. Fibonacci is the hardest case because its high density ($\rho = 0.72$) means the smallest domain L for a given N , and hence the coarsest k -resolution. Silver and Bronze converge at $N \sim 2\text{--}4M$.

Technical Challenges & Solutions

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- **Fix 2:** Linear fit of $\ln E$ vs $\ln t$ over plateau window for single robust α value

Summary of All Results

Real-space analysis (Phases 1–3):

- Variance code validated against Poisson exact result (1.8% error)
- Class I hyperuniformity confirmed: bounded $\sigma^2(R)$ for all chains
- $\bar{\Lambda}$ increases with μ : $0.200 \text{ (Fib)} < 0.250 \text{ (Ag)} < 0.282 \text{ (Br)}$
- Rescaling invariance: substitution vs. projection agree to 0.1%
- Class I \rightarrow II transition demonstrated via non-ideal strip width

Reciprocal-space analysis (Phase 4):

- $\tilde{\chi}_V(k) \sim k^3$ envelope confirmed for all chains
- Spreadability decay: Poisson $t^{-1/2}$, quasicrystal t^{-2} , lattice exp.
- $\alpha = 3.0 \pm 0.05$ **for all three chains** (linear fit method)

$\alpha = 3$ is a **universal property** of all metallic-mean 1D substitution tilings, confirmed both *analytically* (eigenvalue formula) and *numerically* (spreadability).

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Hyperuniformity in 1D Quasicrystals

- └ Summary & Future Work
- └ Summary of All Results

This slide summarizes all findings across four phases.
The key narrative: we approached the same question ($\alpha = 3?$) from two completely independent directions — real-space variance analysis and reciprocal-space spreadability — and both confirm the theoretical prediction. The agreement across three distinct quasicrystal families with different metallic means provides strong evidence for universality.

Summary of All Results

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Planned:

- Construct 2D quasiperiodic tilings via the **Generalized Dual Method (GDM)**:
 - 5-fold Penrose tiling (golden ratio) — $\alpha = 6$ known
 - 8-fold octagonal tiling (silver mean) — α **unknown**
 - Bronze-ratio equivalent — α **unknown**
- Decorate vertices with disks ($\phi_2 = 0.25$)
- Compute 2D radial spectral density and extract α

Open questions:

- 1 Is $\alpha = 6$ universal for all 2D metallic-mean tilings (analogous to $\alpha = 3$ in 1D)?
- 2 Does the eigenvalue formula generalize correctly to 2D inflation matrices?
- 3 How large must 2D systems be for convergence?

2026-02-25

Hyperuniformity in 1D Quasicrystals
└ Summary & Future Work

└ Future Work: Phase 5 — 2D Extensions

The 2D Penrose tiling has $\alpha = 6$ from the literature. The eigenvalue formula for 2D inflation matrices predicts $\alpha = 6$ for Penrose, but the predictions for silver-mean and bronze-mean 2D tilings have not been tested numerically.

Based on the 1D experience, system size requirements may be substantial and non-trivial. The GDM provides a systematic way to generate arbitrary n -fold tilings.

Future Work: Phase 5 — 2D Extensions

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substitution_tilings.py	Chain generation (3 metallic means)
projection_method.py	Cut-and-project from \mathbb{Z}^2
quasicrystal_variance.py	Number variance + $\bar{\Lambda}$
two_phase_media.py	Spectral density + spreadability
run_all.py	Full pipeline: Figs 1–8 + tables

- Reproducibility:** `python run_all.py`
- Deterministic (seed = 42)
 - Runtime: ~18 minutes
 - Generates all 8 figures + summary tables
 - Dependencies: NumPy, Matplotlib, SciPy

2026-02-25

└ Appendix: Code & Reproducibility

All code is self-contained in five Python files. The full pipeline is run with a single command and is fully deterministic.

The bottleneck is the Phase 4 FFT computation for the three quasicrystal chains at $N \sim 10^7$ (about 5 minutes each for Silver and Bronze due to larger FFT sizes).

Appendix: Code & Reproducibility

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