

Ferrous iron partitioning between magnesium silicate perovskite and ferropericlaase and the composition of perovskite in the Earth's lower mantle

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[1] We have investigated the exchange of Fe and Mg between magnesium silicate perovskite (Mg-Pv) and ferropericlaase (Fp) at 25 GPa and 2400 to 2600 K using a Kawai-type multianvil apparatus. Each experiment was performed with coexisting metallic Fe, which buffered the oxygen fugacity at the lowest possible level. As the system was Al-free the presence of metallic Fe ensures low ferric iron (Fe^{3+}) contents in all phases. The results are used to extract thermodynamic data to describe Fe^{2+} -Mg partitioning. A thermodynamic assessment and modeling of the available high-pressure partitioning data indicates that the influence of a Fe-spin transition in Fp on Fe-Mg partitioning may be more subtle than previously proposed. Furthermore, we demonstrate that a comparison between perovskite Fe^{2+} contents predicted by the thermodynamic model and previously reported perovskite analyses can be used to estimate Mg-Pv Fe^{3+} concentrations of both Al-bearing and Al-free phases in the previous studies. These estimates show that the Fe^{3+} content of Al-free Mg-Pv depends strongly on oxygen fugacity, and varies accordingly with the capsule materials used in experiments. The relationship between Fe^{3+} and Al concentrations in Al-bearing Mg-Pv indicates that the substitution mechanism of Fe^{3+} and Al changes with Al content. Chemical heterogeneities in the lower mantle will result in the formation of Mg-Pv with quite different Al and bulk Fe concentrations, which will cause important differences in Fe^{3+} and oxygen vacancy concentrations in Mg-Pv.

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1. Introduction

[2] $(\text{Mg,Fe})\text{SiO}_3$ perovskite (Mg-Pv) and $(\text{Mg,Fe})\text{O}$ ferropericlaase (Fp) comprise more than 90 volume % of the Earth's lower mantle [e.g., *Irfune*, 1994]. Knowledge of the chemical and physical properties of these phases is, therefore, essential for understanding the structure and dynamics of the lower mantle. The partitioning of Fe between the dominant lower mantle phases has been intensively studied because the iron contents of the respective phases will influence lower mantle transport properties such as rheology, elasticity, and electrical and thermal conductivities [e.g., *Mao et al.*, 1997]. The sharpness, i.e., pressure/depth interval, of the post-spinel transition from $(\text{Mg,Fe})_2\text{SiO}_4$ spinel to Mg-Pv and Fp, which is considered to cause the 660 km seismic discontinuity, will depend mainly on the partitioning of Fe-Mg between the

transforming phases [*Ito and Takahashi*, 1989]. Seismic studies argue that this is a remarkably sharp phase transformation occurring over a depth interval corresponding to only a few tenths of a gigapascal (GPa) [e.g., *Benz and Vidale*, 1993]. In addition, changes in the Fe spin state have also been considered to influence Fe-Mg partitioning [e.g., *Badro et al.*, 2003; *Lin et al.*, 2005; *Irfune et al.*, 2010]. In a reciprocal fashion, the onset pressure of the spin transition will also be influenced by the Fe content of the phase [*Fei et al.*, 2007].

[3] The partitioning of Fe-Mg between Mg-Pv and Fp can be described using the exchange, or distribution coefficient, K_D :

$$K_D = \left(X_{\text{FeSiO}_3}^{\text{Pv}} / X_{\text{MgSiO}_3}^{\text{Pv}} \right) / \left(X_{\text{FeO}}^{\text{Fp}} / X_{\text{MgO}}^{\text{Fp}} \right), \quad (1)$$

where X_i^a is the mole fraction of component i in phase a . Ideally K_D describes Fe^{2+} -Mg partitioning where, for example, $X_{\text{FeSiO}_3}^{\text{Pv}}$ is equal to the molar ratio $\text{Fe}^{2+}/(\text{Fe}^{2+} + \text{Mg})$. However, in most studies $\text{Fe}^{3+}/\Sigma\text{Fe}$ ($\Sigma\text{Fe} = \text{Fe}^{2+} + \text{Fe}^{3+}$) ratios of phases have not been determined and an apparent K_D or $K_D(\text{app})$ is actually measured where $X_{\text{FeSiO}_3}^{\text{Pv}}$ is, in reality, the molar ratio $(\text{Fe}^{2+} + \text{Fe}^{3+})/(\text{Fe}^{2+} + \text{Fe}^{3+} + \text{Mg})$.

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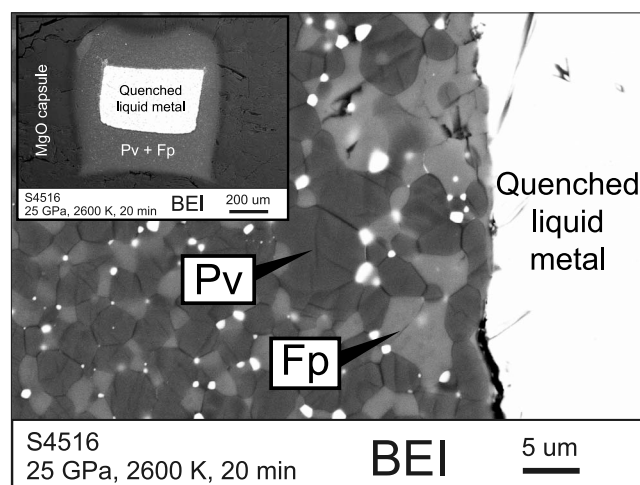


Figure 1. Backscattered electron image (BEI) of a recovered sample that consists of quenched liquid metal coexisting with Mg-perovskite (Pv) and ferropericlasite (Fp). The inset shows a low-magnification image of the entire sample contained in an MgO capsule.

[4] A number of experiments employing both natural mantle compositions and compositions in the simple MgO-FeO-SiO₂ system have been performed using the multianvil apparatus and diamond anvil cells to investigate temperature, pressure, and composition effects on K_D . There are many discrepancies between experimental studies, even in simplified systems, and particularly between studies performed at pressures higher than 25 GPa in the laser-heated diamond anvil cell [Auzende *et al.*, 2008; Sakai *et al.*, 2009; Sinmyo *et al.*, 2008]. A number of possible explanations exist for experimental and analytical discrepancies such as sluggish kinetics of chemical equilibration [Katsura and Ito, 1996; Frost and Langenhorst, 2002] and heterogeneity in the chemical composition and temperature of the samples [Sinmyo *et al.*, 2008]. The valence state of iron in Mg-Pv and Fp will also have a strong effect on K_D , or at least on apparent K_D values derived from electron microprobe and transmitting electron microscopy analyses because the definition of K_D describes only ferrous iron (Fe²⁺) exchange.

[5] The ratio of ferric to ferrous iron in both minerals depends to some extent on oxygen fugacity but for Mg-Pv the Al concentration has by far the most dramatic influence [Frost and Langenhorst, 2002; McCammon *et al.*, 2004]. Ferric iron is favorably partitioned into Mg-Pv, particularly in Al-bearing systems, whereas ferrous iron is preferentially partitioned into Fp. This opposing behavior complicates partitioning determinations as it requires ferrous and ferric Fe to be distinguished in the recovered samples and, for a rigorous description, requires either control or measurement of the experimental oxygen fugacity. Due to these complications some uncertainties in the effects of f_{O_2} and temperature still exist for Fe partitioning behavior between Mg-Pv and Fp, even for Al-free systems at pressures attainable by the large-volume multianvil apparatus. More importantly, partitioning trends determined at relatively well-constrained conditions in multianvil experiments are crucial for evaluating results obtained at higher pressures in the diamond anvil cell [e.g., Auzende *et al.*, 2008; Sinmyo *et al.*, 2008] and for

determining the thermodynamic parameters that are required to extrapolate partitioning behavior to higher pressures and temperatures.

[6] In this study we have examined the exchange coefficient of Fe between Mg-Pv and Fp at low-oxygen fugacity imposed by the coexistence of metallic Fe. These conditions ensure low levels of ferric Fe in each phase over the range of compositions investigated and thus allow Fe²⁺-Mg partitioning to be characterized. Experiments were performed at high temperatures of 2400–2600 K, i.e., close to the silicate solidus, to alleviate kinetic problems observed at lower temperatures in metallic Fe-bearing systems [Frost and Langenhorst, 2002].

2. Experimental and Analytical Procedure

[7] High-pressure experiments were carried out at a pressure of 25 GPa using a Kawai-type multianvil apparatus at the Bayerisches Geoinstitut. In each experiment, a 10-mm MgO octahedral pressure medium was compressed by tungsten-carbide cubes with 4 mm truncated edge lengths (10/4 assembly). A cylindrical LaCrO₃ heater was employed and the temperature was measured using a W3%Re-W25%Re thermocouple. Further experimental details are described by Frost and Langenhorst [2002].

[8] Different compositions of (Mg_{1-x}Fe_x)₂SiO₄ olivine (Fo#70, 90, 95, where Fo# = $(1-x) \times 100$) were prepared as starting compositions. Olivine Fo70 and Fo95 samples were synthesized from reagent grade oxide mixtures using a CO-CO₂ gas mixing furnace at 1200°C and log f_{O_2} = -11.0. Natural powdered San Carlos olivine was employed for the Fo90 composition. Olivine powder was packed together with a piece of 0.5 mm diameter Fe-wire into an MgO single crystal capsule. The samples were compressed to 25 GPa and then annealed at temperatures between 2400 and 2600 K for 10–60 min. The temperature uncertainties are estimated to be less than ± 100 K based on the reproducibility of the temperature-power relationship. The recovered samples were examined by Raman spectroscopy and scanning electron microscope with an electron backscattered diffractometer. Quantitative chemical analyses were performed using an electron microprobe with a wavelength dispersive x-ray spectrometer. Operating conditions were 15 kV and 10 nA with a focused beam. The measurement time for each element was 20–30 s. Mg₂SiO₄ forsterite and metallic Fe were used as internal standards.

3. Results

3.1. Fe-Mg Distribution Coefficients

[9] A backscattered electron image of a typical run product is shown in Figure 1. The recovered samples consisted of Mg-Pv and Fp together with metallic Fe, and no other phases were observed. From textural observations, the metallic iron melted at temperatures above 2350 K. The grain sizes of Mg-Pv and Fp near the liquid metal are large ($>5 \mu\text{m}$) relative to those found further from the metal, implying that liquid metal may act as a catalyst for grain growth. This catalytic effect also appears to promote chemical equilibrium (or at least chemical phase homogeneity) between coexisting Mg-Pv and Fp phases [Katsura and Ito, 1996]. The chemical compositions of Mg-Pv and Fp were obtained from grains

Table 1. Experimental Conditions and Compositions of Perovskite and Ferropericlasel^a

Run	Starting Materials	T (K)	Duration (min)	X_{Fe}^b		K_D^c	ΔIW^d
				Pv	Fp		
S4797	Fo70	2450	10	0.110(3)	0.323(4)	0.26(1)	−0.64(3)
S4882	Fo70	2500	40	0.161(15)	0.487(24)	0.20(3)	−0.43(7)
S4434	Fo90	2430	60	0.071(4)	0.235(9)	0.25(2)	−0.88(8)
S4527	Fo90	2500	10	0.070(4)	0.191(2)	0.32(2)	−1.02(3)
S4796	Fo90	2550	20	0.064(3)	0.188(2)	0.29(2)	−1.03(4)
S4441	Fo90	2573	30	0.073(6)	0.210(3)	0.30(3)	−0.94(3)
S4510	Fo95	2400	10	0.061(2)	0.177(17)	0.30(4)	−1.07(20)
S4545	Fo95	2590	30	0.047(2)	0.119(2)	0.37(2)	−1.38(3)
S4516	Fo95	2600	20	0.058(3)	0.147(4)	0.35(2)	−1.19(6)
S4555	Fo95	2600	30	0.044(3)	0.122(3)	0.33(2)	−1.35(7)

^aNumbers between parentheses are one standard deviation.^bFe/(Fe + Mg)^c $K_D = (X_{\text{Fe}}^{\text{Pv}}/X_{\text{Mg}}^{\text{Pv}})/(X_{\text{Fe}}^{\text{Fp}}/X_{\text{Mg}}^{\text{Fp}})$.^dOxygen fugacity relative to iron-wüstite buffer.

adjacent to liquid metal. Although localized chemical compositional changes occurred as a result of reaction with the metal, the composition of each phase in each experiment was relatively constant in the proximity of the liquid metal, within the uncertainties listed in Table 1.

[10] Compositional data and distribution coefficients are summarized together with the experimental conditions in Table 1. No significant change in K_D was observed between experiments performed at similar conditions for annealing times of 10 and 60 min, implying that for temperatures in the range 2400–2600 K and in the presence of liquid Fe metal, chemical equilibrium is achieved within 10 min. The oxygen fugacity was calculated from Fe and FeO concentrations in metal and ferropericlasel respectively [Mann *et al.*, 2009] and log f_{O_2} values are reported relative to the iron-wüstite oxygen buffer (ΔIW). The oxygen fugacity range in the experiments is between ΔIW -0.4 and ΔIW -1.4, with the lowest oxygen fugacities corresponding to the samples with the lowest bulk FeO contents.

[11] The Mg-Pv and Fp distribution coefficients for Fe-Mg partitioning are plotted in Figure 2 as a function of the mole fraction of FeO in Fp. Iron preferentially partitions into Fp but higher temperatures result in higher values of K_D , i.e., a decrease in the Fe preference for Fp. This trend is even more apparent when the results of Frost and Langenhorst [2002] and McCammon *et al.* [2004] are considered, which were obtained at temperatures <2000 K (Figure 2).

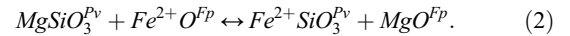
[12] Mössbauer spectroscopy measurements [Frost *et al.*, 2001; McCammon *et al.*, 2004] indicate that Fe^{3+} concentrations in ferropericlasel coexisting with metallic iron at high pressures, >18 GPa, are negligible for Fe/(Fe + Mg) ratios <0.5 (Figure 3). It should be noted, however, that although the Fp Fe^{3+} contents are negligible for Fe/(Fe + Mg) ratios obtained in the current study, the $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratio of Fp will be small but significant (2–5%) for Fe/(Fe + Mg) ratios >0.8 [McCammon, 1992; Frost *et al.*, 2001].

[13] For Mg-Pv in the Al-free system, the dominant influences on the Fe^{3+} content are the total Fe concentration and oxygen fugacity [McCammon *et al.*, 1997, 2004]. For Al-free (Mg,Fe)SiO₃ perovskite coexisting with metallic Fe with Fe/(Fe + Mg) ratios <0.2, $\text{Fe}^{3+}/\Sigma\text{Fe}$ values in the range 0.07 to 0.1 have been reported [Lauterbach *et al.*, 2000; McCammon *et al.*, 1997]. In the present study the

Fe/(Mg + Fe) ratios of Mg-Pv lie in the range 0.04–0.15 and the Fe^{3+} concentrations will therefore be at a similar level to the analysis errors (Table 1). As a result of buffering Fe^{3+} levels at the lowest practical level we can assume that the partitioning results dominantly describe Fe^{2+} -Mg partitioning with a negligible effect of Fe^{3+} such that the apparent K_D or $K_D(\text{app}) \approx K_D$.

3.2. Thermodynamics of Fe-Mg Partitioning Between Perovskite and Ferropericlasel

[14] The Fe^{2+} and Mg exchange equilibrium between Mg-Pv and Fp is



The distribution coefficient K_D is then defined as in equation (1). At equilibrium K_D can be described by the expression:

$$RT \ln K_D = -\Delta G_{p,T}^0 - W_{\text{FeMg}}^{\text{Pv}} (1 - 2X_{\text{FeSiO}_3}^{\text{Pv}}) + W_{\text{FeMg}}^{\text{Fp}} (1 - 2X_{\text{FeO}}^{\text{Fp}}), \quad (3)$$

where R is the gas constant, $\Delta G_{p,T}^0$ is the standard state Gibbs free energy change for the pure end-member equilibrium at the pressure and temperature of interest and $W_{\text{FeMg}}^{\text{Pv}}$ and $W_{\text{FeMg}}^{\text{Fp}}$ are symmetric interaction parameters (Margules parameters) for Mg-Pv and Fp that account for the non-ideality of Fe-Mg mixing in each solid solution. The standard state Gibbs free energy change for equilibrium (2) can be simply described by

$$\Delta G_{p,T}^0 = \Delta H^0 + P\Delta V^0 - T\Delta S^0, \quad (4)$$

where ΔH^0 , ΔV^0 , and ΔS^0 are changes in enthalpy, volume, and entropy, respectively. Equations (3) and (4) can be fit to the experimental data and to the results of Frost and Langenhorst [2002]. However, as the two interaction parameters are highly correlated it is more practical to fix one of these parameters using results of previous studies. The Margules parameter $W_{\text{FeMg}}^{\text{Fp}}$ for ferropericlasel has been well

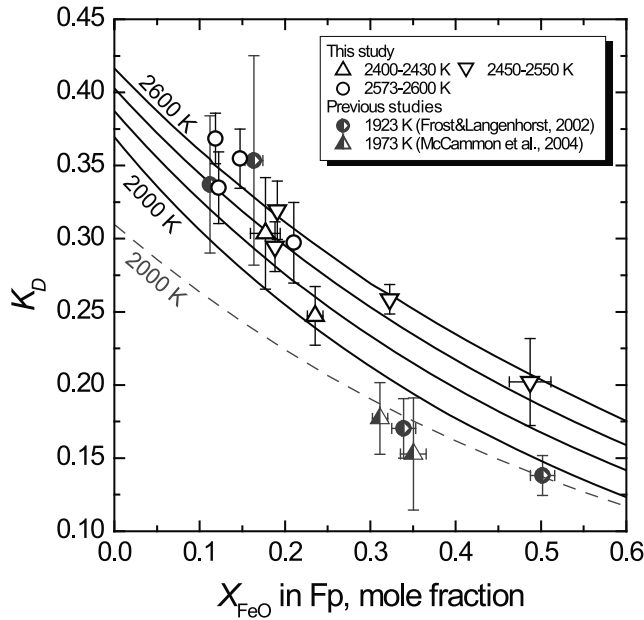


Figure 2. Experimental results of Fe-Mg partitioning between Pv and Fp plotted as K_D against X_{FeO} in Fp. Results from the previous studies of *Frost and Langenhorst* [2002] and *McCammon et al.* [2004] are also shown. The solid lines show isothermal K_D values (200 K intervals) that result from fitting equation (3) to the data. The broken line shows K_D calculated from the thermodynamic parameters proposed by *McCammon et al.* [2004].

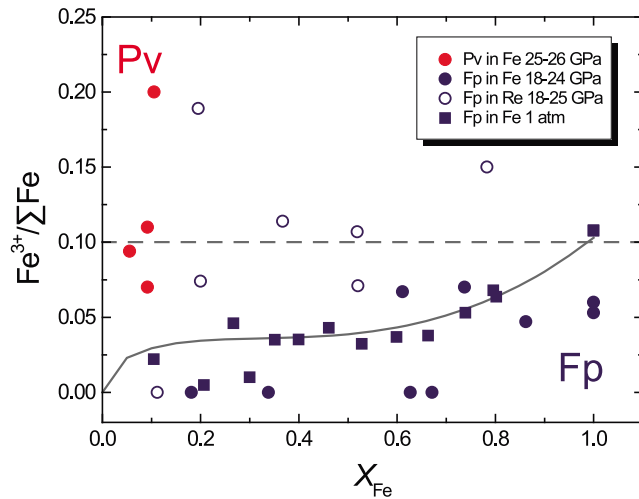


Figure 3. Ferric iron contents of perovskite and ferropericlasite. Values of ferric iron relative to total iron content ($\text{Fe}^{3+}/\Sigma\text{Fe}$) in perovskite (Pv – red symbols) and ferropericlasite (Fp – blue symbols) are plotted as a function of X_{FeO} . Perovskite data are from *Lauterbach et al.* [2000] and *McCammon et al.* [1997]. Data for ferropericlasite coexisting with metallic Fe were obtained at 1 bar [*Katsura and Kimura*, 1965; *O'Neill et al.*, 2003; *Srećec et al.*, 1987] and at 18–24 GPa [*Frost et al.*, 2001]. Data for ferropericlasite contained in Re capsules are from *Frost et al.* [2001] and *Frost and Langenhorst* [2002]. Solid and broken lines show calculated $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratios of Fp coexisting with metallic Fe [*O'Neill et al.*, 2003] and Re (this study), respectively.

studied over a wide pressure, temperature and compositional range [*Frost*, 2003] and $W_{\text{FeMg}}^{\text{Fp}} = 14$ kJ/mol at 25 GPa is taken from this previous work. By fitting to the 25 GPa results the following terms are derived: $\Delta H^0 + P\Delta V^0 = 27.2 \pm 6.6$ kJ/mol, $\Delta S^0 = -4.0 \pm 1.9$ J/mol K⁻¹, and $W_{\text{FeMg}}^{\text{Pv}} = -4.9 \pm 7.6$ kJ/mol. Because the compositional range of $X_{\text{FeSiO}_3}^{\text{Pv}} = 0.04\text{--}0.15$ covered by this study is small, the uncertainty of $W_{\text{FeMg}}^{\text{Pv}}$ is relatively large and fixing $W_{\text{FeMg}}^{\text{Pv}}$ to be 0 kJ/mol is equally justified. The expression for $\Delta G_{\text{P,T}}^0$ then becomes $\Delta H^0 + P\Delta V^0 = 24.1 \pm 4.4$ kJ/mol and $\Delta S^0 = -3.5 \pm 1.7$ J/mol K⁻¹. The fitted curves calculated using obtained parameters with $W_{\text{FeMg}}^{\text{Pv}} = -4.9$ kJ/mol are shown in Figure 2.

4. Discussion

4.1. Comparison With Previous Thermodynamic Models and Calculating the Depth Interval of the 660 km Seismic Discontinuity

[15] Using the present experimental data and fixing interaction parameters for Mg-Pv and Fp as given in the last section, we have calculated the standard state Gibbs free energy change $\Delta G_{\text{P,T}}^0$ for equilibrium (2) at 25 GPa from equation (3) (Figure 4). The uncertainties in $\Delta G_{\text{P,T}}^0$ were propagated mainly from the uncertainty in the determination of $W_{\text{FeMg}}^{\text{Pv}}$. The effect of temperature uncertainty on the $\Delta G_{\text{P,T}}^0$ calculation is negligible, even when taking the maximum uncertainty (100 K) of the present temperature measurements into

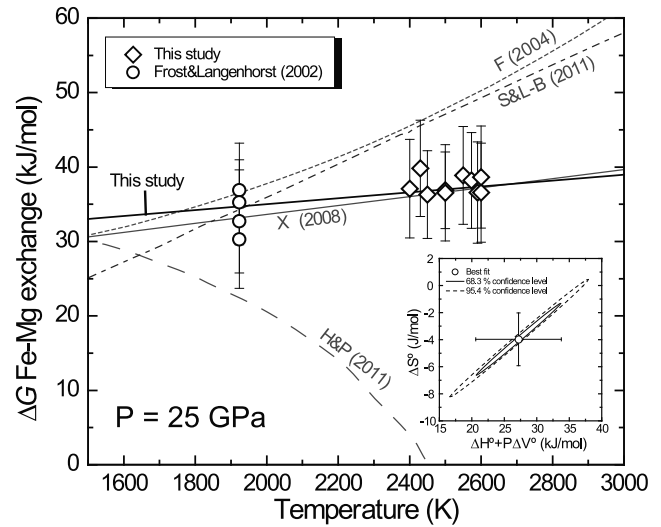


Figure 4. The standard state Gibbs free energy change ($\Delta G_{\text{P,T}}^0$) of the Fe^{2+} -Mg exchange reaction between perovskite and ferropericlasite at 25 GPa as a function of temperature. The Gibbs free energies were obtained from isothermal fits of equation (3) to the present results and those of previous studies [*Frost and Langenhorst*, 2002]. The black line shows the best fit to these data. The labeled gray lines are from thermodynamic models by *Fabrichnaya et al.* [2004]: F (2004), *Holland and Powell* [2011]: H&P (2011), *Stixrude and Lithgow-Bertelloni* [2011]: S&L-B (2011), and *Xu et al.* [2008]: X (2008). The inset shows the confidence ellipsoid from the determination of the entropy change (ΔS^0) and enthalpy and volume change ($\Delta H^0 + P\Delta V^0$).

Table 2. Properties of End-Member Components for the Helmholtz Free Energy Calculation^a

	F_0 (kJ/mol)	V_0 (cc/mol)	K_0 (GPa)	K'_0	θ_0 (K)	γ_0	q_0	References
MgSiO ₃ -Pv	-1368	24.45	251	4.1	905	1.57	1.1	<i>Xu et al.</i> [2008]
FeSiO ₃ -Pv	-1018	25.40	272	4.1	765	1.44	1.4	<i>Xu et al.</i> [2008]
FeSiO ₃ -Pv ^b	-1023	25.34	272	4.1	765	1.44	1.4	<i>Xu et al.</i> [2008], <i>Tange et al.</i> [2009], this study
MgO	-570	11.24	161	3.9	772	1.48	1.6	<i>Xu et al.</i> [2008]
FeO	-244	12.26	179	4.9	454	1.54	1.6	<i>Xu et al.</i> [2008]
FeO ^b	-244	12.26	149.4	3.6	417	1.41	0.5	<i>Xu et al.</i> [2008], <i>Fischer et al.</i> [2011]

^aHelmholtz free energy, F_0 , volume, V_0 , bulk modulus, K_0 , pressure derivative of the bulk modulus, K'_0 , Debye temperature, θ_0 , Grüneisen parameter, γ_0 , logarithmic volume derivative of the Grüneisen parameter, q_0 , at ambient conditions.

^bZero pressure volume of FeSiO₃-Pv and thermoelastic parameters of FeO are taken from recent experimental studies instead of the original values used in *Xu et al.* [2008].

account. The $\Delta G_{P,T}^0$ values, calculated by also employing the results of previous experiments [*Frost and Langenhorst*, 2002], indicate a weak dependence on temperature, corresponding to an entropy change (ΔS^0) of -4.0 J/mol K. The insert of Figure 4 shows confidence ellipsoids for the fitted ΔS^0 and $\Delta H^0 + P\Delta V^0$ values, indicating that the uncertainties on each term are highly correlated.

[16] This determination of $\Delta G_{P,T}^0$ provides the only rigorous approach for approximating the Gibbs free energy for the FeSiO₃ perovskite end-member because all other end-members in equilibrium (2) can be estimated independently. Similarly these results can be used to test predictions of existing thermodynamic databases for the calculation of $\Delta G_{P,T}^0$ as shown in Figure 4. Results show that several previous studies provide poor estimates for $\Delta G_{P,T}^0$ particularly at high temperature. Differences between the experimental results and thermodynamic calculations of *Fabrichnaya et al.* [2004], *Holland and Powell* [2011], and *Stixrude and Lithgow-Bertelloni* [2011] become large at temperatures >2000 K, but excellent agreement is found with the study of *Xu et al.* [2008]. Since lower mantle temperatures lie in the range 1900–2700 K [e.g., *Stacey and Davis*, 2004], reliable thermodynamic data at high temperatures are critical for modeling the lower mantle.

[17] The ability to accurately determine $\Delta G_{P,T}^0$ for equilibrium (2) is essential for calculating the pressure or depth interval of the (Mg_{0.9}Fe_{0.1})₂SiO₄ ringwoodite to Mg-Pv + Fp transformation, which is considered to cause the 660 km seismic discontinuity. As the transformation is divariant and Fe and Mg partition differently between the three phases, it occurs over a specific pressure interval within which all three phases coexist. Seismology can provide a qualitative estimate for the depth interval of the corresponding jump in seismic velocities in the Earth. Using data consistent with Figure 4 and ringwoodite Fe-Mg mixing parameters from *Frost et al.* [2001] this pressure interval can be calculated to be 0.025 GPa at 1873 K, which corresponds to a depth interval in the Earth of less than 1 km. Such a narrow interval would be impossible to determine through phase equilibrium experiments. Although the presence of Al in the Earth may serve to broaden this transition to some extent, experimental studies have shown relatively small Al concentrations in Mg-Pv at the pressure of the transformation [*Hirose*, 2002]. Such a sharp depth interval is in excellent agreement with seismic observations [e.g., *Benz and Vidale*, 1993] that indicate a sharp 660 km discontinuity capable of reflecting short period seismic waves.

4.2. Effect of Pressure on Fe-Mg Partitioning

[18] The effect of pressure on K_D has been intensely investigated through diamond anvil cell (DAC) experiments. There are, however, significant inconsistencies between some studies, and even between those performed using the same starting material (e.g., natural San Carlos olivine) and under similar experimental conditions [*Auzende et al.*, 2008; *Kobayashi et al.*, 2005; *Sakai et al.*, 2009; *Sinmyo et al.*, 2008]. Using the data from this study, the temperature and compositional effects on Fe-Mg partitioning can be described and by making some assumptions about the volume change of equilibrium (2), an assessment of the likely effect of pressure can be made. This is important because without some basic understanding of the plausible magnitude of the pressure dependence, anomalous behavior that might arise from changes in the Fe-spin state in either phase cannot be recognized.

[19] Two methods are used here to assess the likely pressure dependence of Fe-Mg partitioning. The first uses thermodynamic data on Mg-Pv and Fp, extrapolated using an equation of state model, while the second uses the high-pressure data themselves to assess the volume change implied by each experimental data point.

[20] For the first method of extrapolation, independent estimates of the volumetric and equation of state (EoS) properties of the components in equilibrium (2) are used in the EoS model of *Xu et al.* [2008] to calculate values of K_D at high pressure. The calculation is made for a (Mg_{0.9}Fe_{0.1})₂SiO₄ composition at 2000 K, i.e., similar to most DAC samples [*Auzende et al.*, 2008; *Sakai et al.*, 2009; *Sinmyo et al.*, 2008]. The thermodynamic parameters used in the calculation are given in Table 2. K_D values were calculated using the model of *Xu et al.* to determine $\Delta G_{P,T}^0$ for equilibrium (2) at high P - T conditions and using the Margules parameters W_{FeMg}^{Pv} and W_{FeMg}^{Fp} obtained in this study (see section 3.2). The effect of pressure on W_{FeMg}^{Fp} was taken from *Frost* [2003], whereas that on W_{FeMg}^{Pv} was considered to be zero in the absence of evidence to the contrary. As can be seen in Figure 5, K_D values predicted using this approach are considerably higher than the results of most experimental studies. The differences between the calculated K_D and experimental data become significant with increasing pressure and it is clear that the EoS model of *Xu et al.* does not reproduce the volume change of equilibrium (2). This inconsistency might be alleviated through the selection of different EoS parameters for the various components with terms for FeSiO₃-Pv and FeO being the most uncertain.

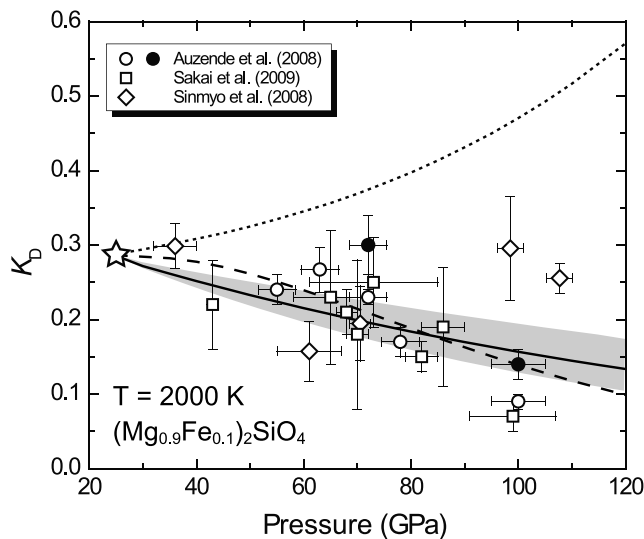


Figure 5. The effect of pressure on Fe-Mg partitioning (K_D). The solid line shows the K_D for a $(\text{Mg}_{0.9}, \text{Fe}_{0.1})_2\text{SiO}_4$ composition at 2000 K calculated from equation (6) by assuming the average volume change ($0.20 \pm 0.04 \text{ cm}^3/\text{mol}$) for the ion exchange equilibrium determined by considering all available DAC experimental data (see Figure 6 and text for details). The gray region shows the uncertainty on this K_D estimate propagated from the resultant volume change uncertainty. The dotted line is calculated from the thermodynamic model of *Xu et al.* [2008]. The dashed line is also calculated using the same model of *Xu et al.* [2008] but with the EoS data for FeO replaced by those from the recent experimental study of *Fischer et al.* [2011] and the ambient volume of FeSiO_3 replaced by that of *Tange et al.* [2009]. These K_D calculations are compared with previous DAC experiments performed with similar starting compositions and at temperature conditions of 2000–2450 K [Auzende et al., 2008], 1860–2120 K [Sakai et al., 2009] and 1760–2170 K [Sinmyo et al., 2008]. Open and filled symbols show transmission electron microscopy (TEM) and secondary ion mass spectroscopy (SIMS) measurements, respectively.

The FeO EoS employed by *Xu et al.* [2008], for example, predicts relatively large volumes at high pressure, compared to the recent experimental study of *Fischer et al.* [2011]. We found that modification of the zero-pressure volume of FeSiO_3 -Pv and the EoS of FeO based on the recent studies of *Tange et al.* [2009] and *Fischer et al.* [2011], respectively, lead to calculated K_D values which are in much better agreement with the experimental DAC partition data. The K_D calculated using these modified EoS parameters (Table 2) is shown by the dashed line in Figure 5.

[21] The second approach is to use equations (3) and (4) to calculate K_D , where the pressure effect is simply described by a constant volume change ΔV^0 of reaction (2) and the remaining terms are determined using the results from this study. ΔV^0 can be evaluated using each experimental data point from the previous DAC experiments performed over the entire lower-mantle pressure range [Auzende et al., 2008; Kobayashi et al., 2005; Sakai et al., 2009; Sinmyo et al., 2008]. From the $K_D(P, T)$ obtained at a pressure (P) and temperature (T) in each experiment and $K_D(P_0, T)$ calculated

from thermodynamic parameters determined at $P_0 = 25 \text{ GPa}$ in this study, ΔV^0 for each experimental result is calculated from

$$\Delta V^0 = RT \ln[K_D(P, T, X)/K_D(P_0, T, X)]/(P_0 - P). \quad (5)$$

The calculated ΔV^0 value for each experiment is shown in Figure 6. Uncertainties are estimated by propagating the uncertainties in K_D and pressure as reported in the literature. Calculated ΔV^0 values are positive and almost constant from 40 to 110 GPa with an average value of $0.20 \pm 0.04 \text{ cm}^3/\text{mol}$. Below 40 GPa, this calculation has large uncertainties due to the small deviations of pressure from $P_0 = 25 \text{ GPa}$. The ΔV^0 values for multianvil experiments performed to 50 GPa by *Tange et al.* [2009] are also calculated (Figure 6) and result in average volumes that are consistent with other studies, even though a very Fe-rich bulk composition was employed. Figure 6 therefore provides an effective way of comparing the effect of pressure alone on partitioning because variations in Fe content and temperature should be effectively normalized in this analysis.

[22] Using the average value of ΔV^0 , the enthalpy change ΔH^0 for reaction (2) is further refined to be 22.3 kJ/mol. With the obtained ΔV^0 and ΔH^0 values, equation (3) becomes

$$RT \ln K_D = -22300 - 200P - 4.0T + 4900(1 - 2X_{\text{FeSiO}_3}^{\text{Pv}}) + (11000 + 110P)(1 - 2X_{\text{FeO}}^{\text{Fp}}), \quad (6)$$

where pressure (P) is in GPa and temperature (T) is in K.

[23] Using equation (6), K_D for the $(\text{Mg}_{0.9}, \text{Fe}_{0.1})_2\text{SiO}_4$ composition at 2000 K is calculated up to 120 GPa (solid line in Figure 5). At 100 GPa, K_D is calculated to be 0.16 with an uncertainty (shaded region in Figure 5) of ± 0.03 based on the volume change error.

[24] It should be noted that some of the discrepancies in K_D , measured in DAC investigations [Auzende et al., 2008; Sakai et al. 2009; Sinmyo et al., 2008], result from variations in the bulk Fe contents and temperatures of individual studies. These discrepancies appear much smaller, however, when the actual Fe contents of the phases involved are considered. The model predicts a change in the Mg-Pv Fe/(Fe + Mg) ratio for a $(\text{Mg}_{0.9}, \text{Fe}_{0.1})_2\text{SiO}_4$ bulk composition from 0.048 at 25 GPa to 0.031 at 100 GPa, a relatively small shift for a large pressure change. At 100 GPa conflicting experimental Mg-Pv Fe/(Fe + Mg) ratios lie in the range 0.026–0.036, which is also very small, particularly when compared to the analytical uncertainties of at least ± 0.01 in this ratio as determined from results of DAC studies. Therefore, although experimental K_D values seem to be quite widely scattered in Figure 5 the actual differences in Fe contents between phases at similar conditions are relatively small and calculating K_D tends to exaggerate these small differences.

[25] *Auzende et al.* [2008] and *Sakai et al.* [2009] reported an abrupt reduction in K_D at around 70–100 GPa and attributed this to the Fe^{2+} electronic spin transition from high-spin to low-spin in Fp. As shown in Figure 6 there is no consistent shift in the determined volume change of equilibrium (2), even within results from a single study. It is therefore hard

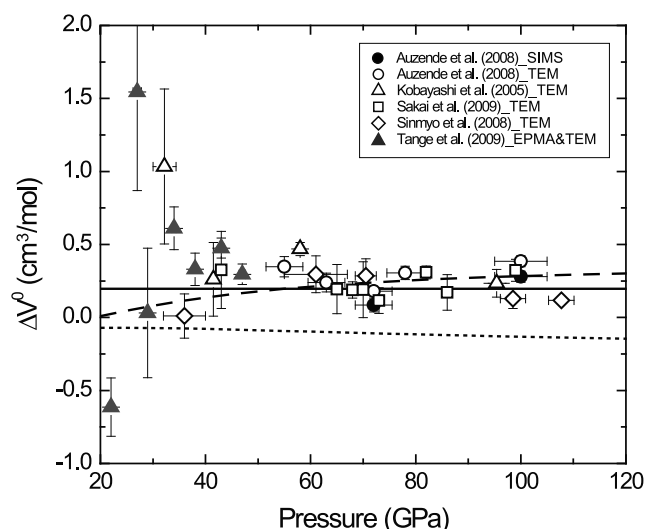


Figure 6. The standard state volume change for the Fe-Mg exchange reaction between perovskite and ferropericlasite as a function of pressure. The volumes are estimated from individual DAC and multianvil experimental partitioning results using the thermodynamic model derived in this study at 25 GPa (see text for details). The solid line shows the average volume change of 0.20 ± 0.02 cm³/mol obtained from a fit to the data (see text for details). The dotted line is calculated from the thermodynamic parameters given in Xu *et al.* [2008]. The dashed line is calculated from same model modified with the recent experimental results by Tange *et al.* [2009] and Fischer *et al.* [2011], as described for Figure 5.

to argue that the DAC results cannot be explained by a single value of ΔV^0 or a value changing only as a response to elastic compression. In fact, it is doubtful that the spin transition in Fp can cause abrupt changes in partitioning behavior in this system. Experimental and theoretical studies show that the Fe²⁺ spin state in Fp changes gradually over a broad pressure range at high temperature (e.g., over the range 50–90 GPa at 2000 K for Mg_{0.75}Fe_{0.25}O) [Lin *et al.*, 2007; Tsuchiya *et al.*, 2006]. Moreover, Li [2007] suggested from calculations based on crystal field theory that K_D would not change abruptly due to the spin transition even if it did occur sharply over a narrow pressure range.

[26] There are several possible additional reasons for the scatter in the results of the DAC studies. (1) Sinmyo *et al.* [2008] reported that large heterogeneities in the compositions of Mg-Pv and Fp in recovered DAC samples are caused by large temperature gradients across the samples during laser heating. This can potentially result in erroneous partition coefficients that are simply transient as Fe diffuses in or out of each phase at different rates. (2) Fe-Mg diffusion in Mg-Pv is extremely sluggish, even at high temperatures [Holzapfel *et al.*, 2005] so that equilibration may not always be achieved. For these reasons it is crucial that DAC studies are supported by thermodynamic calculations based on measurements of properties that can be obtained independently and with smaller realistic uncertainties than DAC chemical partitioning data. Effects on K_D from Fe-spin transitions are only likely to show up once such thermodynamic analyses are performed and may be too gradual to cause

clearly detectable changes in K_D over narrow pressure intervals.

4.3. Fe³⁺ Concentrations in Al-Free and Al-Bearing Perovskite

[27] $K_D(app)$ (i.e., determined from analyses in which Fe = Fe²⁺ + Fe³⁺) has been found to increase strongly with the Al content of Mg-Pv due to a strong coupling between Al and Fe³⁺ substitution in Mg-Pv, although in detail the relationship is not linear [Wood and Rubie, 1996; Frost and Langenhorst, 2002; McCammon *et al.*, 2004]. Previous studies indicate that the presence of Al and Fe³⁺ in Mg-Pv does not significantly influence Fe²⁺-Mg partitioning between Mg-Pv and Fp and when the Fe³⁺ content is measured and subtracted from the bulk perovskite Fe content, the actual K_D is, within error, the same as that determined in the Al-free system [Frost and Langenhorst, 2002; McCammon *et al.*, 2004]. Therefore, if the Fe²⁺ content of Mg-Pv can be estimated from the corresponding Fe content of Fp, using determinations of K_D , then any additional Fe found in Mg-Pv can be assigned to be Fe³⁺. In this way the Fe³⁺ contents of Mg-Pv can be determined from chemical analyses in the literature where no explicit measurement of the Fe³⁺/ΣFe ratio has been made.

[28] Based on chemical analyses of phases from previous experimental studies, the Fe²⁺ contents of Mg-Pv were calculated from the corresponding Fe contents of Fp and appropriate K_D values for the *P-T* conditions of each experiment, using the thermodynamic parameters obtained in the current study. The Fe³⁺ contents of Mg-Pv were then calculated as the difference between the determined Fe²⁺ contents and the reported total Fe contents in Mg-Pv (Figure 7). In the calculations, Fe³⁺ concentrations in Fp were assumed to be negligible for experiments performed in Fe and diamond capsules but for Re capsules a Fe³⁺/ΣFe ratio of 0.1 was assumed (Figure 3). In this way Mg-Pv analyses from previous experimental studies performed at 21–30 GPa and 1573–2473 K have been evaluated. Experiments performed at temperatures <2273 K for time scales <1 h were excluded from this treatment on the grounds that chemical equilibrium is unlikely to have been achieved [Katsura and Ito, 1996; Frost and Langenhorst, 2002].

[29] The calculated Fe³⁺ contents, $X_{Fe^{3+}}$ (= Fe³⁺/Fe²⁺ + Fe³⁺ + Mg), of Mg-Pv coexisting with Fp synthesized in several different capsule materials are plotted as a function of the total Fe content of Mg-Pv in Figure 7. Based on the determined uncertainties of the thermodynamic parameters and their interdependencies (inset of Figure 4), the typical uncertainty of calculated Fe³⁺ contents is ± 0.004 . Clearly defined trends are observed as a function of different capsule materials and the presence of Al in the samples. Along each trend, there is good agreement between samples where Fe³⁺/ΣFe ratios have been measured and those that have been estimated using the method described above. For Al-free samples the Mg-Pv Fe³⁺/ΣFe ratios calculated for samples equilibrated in rhenium capsules [Katsura and Ito, 1996; Martinez *et al.*, 1997; Frost and Langenhorst, 2002; McCammon *et al.*, 2004] are in the range 0.1–0.3, which is in good agreement with results from ELNES and Mössbauer spectroscopy measurements [Frost and Langenhorst, 2002; McCammon *et al.*, 2004]. Lower Fe³⁺/ΣFe values are estimated for Al-free Mg-Pv samples synthesized in diamond

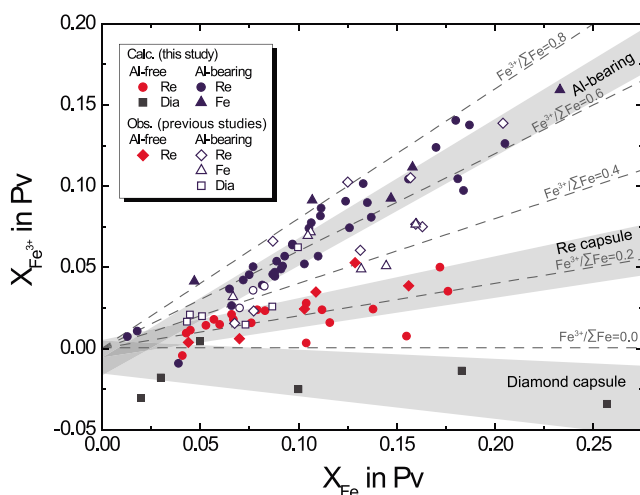


Figure 7. Calculated Fe^{3+} contents of Mg-perovskite coexisting with ferropiclsite in Al-free and Al-bearing systems. $X_{\text{Fe}^{3+}} (= \text{Fe}^{3+}/\text{Fe}^{2+} + \text{Fe}^{3+} + \text{Mg})$ values are plotted as a function of $X_{\text{Fe}} (= [\text{Fe}^{2+} + \text{Fe}^{3+}]/[\text{Fe}^{2+} + \text{Fe}^{3+} + \text{Mg}])$. Mg-Pv Fe^{3+} concentrations were calculated from results of previous experiments performed on Al-free and Al-bearing systems using a variety of capsule materials (see text for details). Black squares and filled red circles show the results of experiments on the Al-free system using diamond [Tange et al., 2009; Yamazaki et al., 2009] and rhenium (Re) capsules [Frost and Langenhorst, 2002; Katsura and Ito, 1996; Martinez et al., 1997], respectively. Solid blue symbols show results for Al-bearing samples contained in Re and metallic iron (Fe) capsules [Frost and Langenhorst, 2002; Hirose et al., 2001; Nishiyama and Yagi, 2003; Wood, 2000]. Open blue symbols and red diamonds show results based on direct measurements of Fe^{3+} in Pv [Frost and Langenhorst, 2002; Frost et al., 2004; Irifune et al., 2010; McCammon et al., 2004]. Broken lines show ferric iron ratios $\text{Fe}^{3+}/(\text{Fe}^{2+} + \text{Fe}^{3+})$ in Mg-Pv. Shaded areas indicate uncertainties on linear fits to data obtained in different capsules and for Al-bearing samples.

capsules [Tange et al., 2009; Yamazaki et al., 2009] compared to those in rhenium capsules. This is in good agreement with the expected oxygen fugacity range for the different capsule types, as experiments in diamond capsules are constrained to be at least 4 orders of magnitude lower in f_{O_2} compared to the maximum f_{O_2} obtainable in Re capsules at 25 GPa and 2000 K [Pownceby and O'Neill, 1994; Stagno and Frost, 2010]. Even though the capsule material should only influence the upper f_{O_2} limit, it seems that, at least for Al-free experiments, the capsule material employed has a clear influence on the f_{O_2} and corresponding Fe^{3+} concentrations. This may be the result of the capsule environment in the multianvil being intrinsically oxidizing as a result of the routine use of LaCrO_3 furnaces, such that all capsule materials become oxidized to some extent and metals such as Re likely form an oxide buffer. $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratios that are estimated to be lower than zero in Figure 7 most likely arise due to disequilibrium as Fe-Mg equilibration between Mg-Pv and Fp is extremely sluggish at low f_{O_2} and temperatures <2200 K [Frost and Langenhorst, 2002].

[30] In Al-bearing systems, $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratios are significantly higher than those in the Al-free system and cover a broader range of values, from 0.2 to 0.8. Figure 8 shows the relationship between Fe^{3+} and Al concentrations in Mg-Pv with values calculated from previous studies [Frost and Langenhorst, 2002; Hirose et al., 2001; Nishiyama and Yagi, 2003; Wood, 2000] compared with observed values where $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratios were actually measured [Frost and Langenhorst, 2002; Frost et al., 2004; Irifune et al., 2010; Lauterbach et al., 2000; Li et al., 2004; Miyajima et al., 2004; McCammon et al., 2004; Saikia et al., 2009]. The agreement between measured and estimated values is good and it can be seen that there is no clear trend separating experiments performed in Re, Pt, Au, diamond or Fe capsules, which is also apparent in Figure 7.

[31] The relationship between Al and Fe^{3+} in Mg-Pv has been shown to be nonlinear [Frost and Langenhorst, 2002], which is likely related to the changing mechanism of 3+ cation incorporation. However, in Figure 8 the data have also been contoured as a function of bulk Fe concentration and it is clear that the Fe^{3+} content also depends relatively linearly on bulk Fe.

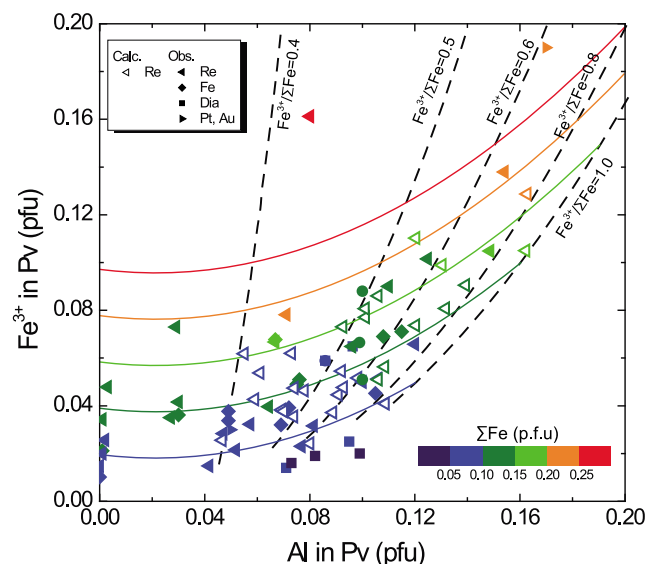


Figure 8. The relationship between Fe^{3+} and Al concentration in Mg-perovskite in atoms per formula unit (pfu) normalized to two cations. The color variation indicates the bulk Fe content in Mg-perovskite (Mg-Pv). Open symbols show Fe^{3+} contents of Mg-Pv calculated from previous studies [Frost and Langenhorst, 2002; Hirose et al., 2001; Nishiyama and Yagi, 2003; Wood, 2000] using the present thermodynamic model (see text). Filled symbols show measurements of Fe^{3+} in Mg-Pv synthesized in capsules consisting of noble metals (Re, Pt, Au), Fe metal, and diamond [Frost and Langenhorst, 2002; Frost et al., 2004; Irifune et al., 2010; Lauterbach et al., 2000; Li et al., 2004; Miyajima et al., 2004; McCammon et al., 2004; Saikia et al., 2009]. Solid lines are fitted curves for different total Fe ($\Sigma\text{Fe} = 0.05\text{--}0.25$) and Al contents with the form: $x_{\text{Fe}^{3+}} = \alpha_0 x_{\text{Al}} + \alpha_1 x_{\text{Al}}^2 + \beta_0 x_{\Sigma\text{Fe}}$. Broken lines indicate $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratio contours.

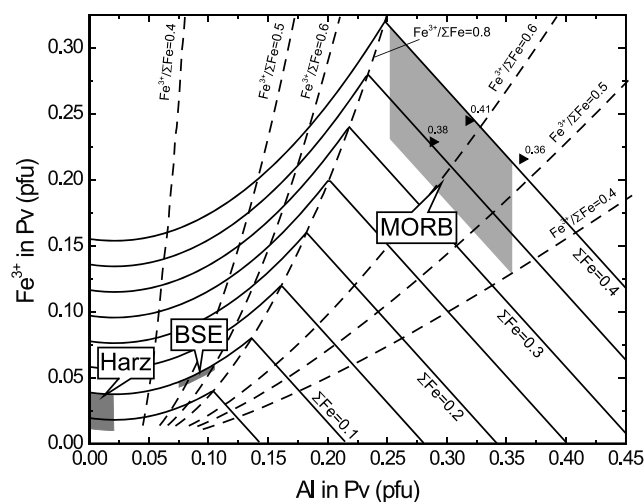


Figure 9. Fe^{3+} and Al concentrations for ranges of plausible lower mantle Mg-perovskite compositions. Broken lines are contours of $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratio and solid lines give Fe^{3+} and Al concentrations for different bulk iron (ΣFe) contents. The maximum value of $\text{Fe}^{3+}/\Sigma\text{Fe}$ is assumed to be 0.8, in line with experimental observations. The Fe^{3+} contents increase with Al content up to $\text{Fe}^{3+}/\Sigma\text{Fe} = 0.8$, and decrease linearly with Al content beyond this maximum ratio, as discussed in the text. Symbols with corresponding total ΣFe values indicated are from measurements by *Litasov et al.* [2003]. The ranges of Fe^{3+} and Al contents of Mg-Pv are indicated for peridotitic or bulk silicate Earth (BSE), harzburgitic (Harz), and basaltic (MORB) compositions by gray shaded regions based on the results of previous studies [*Hirose et al.*, 1999; *Irifune*, 1994; *Irifune and Ringwood*, 1987; *Ono et al.*, 2001].

[32] Two possible mechanisms can be considered for Fe^{3+} and Al^{3+} incorporation into a $\text{Mg}^{2+}\text{Si}^{4+}\text{O}_3$ perovskite structure [e.g., *McCammon*, 1998; *Frost and Langenhorst*, 2002] in which two distinct sites are the pseudo-dodecahedral A-site for 2+ cations and the octahedral B-site for Si^{4+} [e.g., *Horiuchi et al.*, 1987]. One is a coupled substitution mechanism by which both the A- and B-sites are occupied by 3+ cations along the compositional join $\text{MgSiO}_3\text{-Fe}^{3+}\text{Al}^{3+}\text{O}_3$. In this substitution, Fe^{3+} occupies the A-site and is charge balanced by Al^{3+} located on the octahedral site, as determined for Al-rich Fe-bearing Mg-Pv (Al = 0.08 per formula unit, pfu) [*Miyajima et al.*, 2004]. The second is an oxygen vacancy mechanism by which 3+ cations occupy only B-sites along the join $\text{Mg}^{2+}\text{Si}^{4+}\text{O}_3\text{-Mg}^{2+}\text{Fe}^{3+}\text{O}_{2.5}$ (or $\text{Mg}^{2+}\text{Al}^{3+}\text{O}_{2.5}$) and charge balance is achieved by oxygen vacancy formation. The stoichiometry determined from chemical analyses of some Al^{3+} -poor Fe-bearing Mg-Pv is consistent with the oxygen defect mechanism [e.g., *Lauterbach et al.*, 2000; *Frost and Langenhorst*, 2002]. In Al-poor Fe-bearing Mg-Pv (Al = 0.03 pfu), Mössbauer measurements indicate that tetrahedrally coordinated Fe^{3+} might be located on the normally octahedrally coordinated Si site [*Lauterbach et al.*, 2000], which would be consistent with the existence of local oxygen-vacancies. This site occupancy of Fe^{3+} is also observed in perovskites in the system $\text{CaTiO}_3\text{-Ca}^{2+}\text{Fe}^{3+}\text{O}_{2.5}$ [e.g., *Becerro et al.*, 1999].

A strong linear relationship between Al and Fe^{3+} concentrations in Mg-Pv is expected for the coupled substitution mechanism while for the oxygen vacancy mechanism the relationship would likely be weaker [*Frost and Langenhorst*, 2002].

[33] For Al contents <0.08 pfu, the Fe^{3+} content varies mainly with bulk Fe and only weakly with Al. This is most likely due to the dominance of an oxygen vacancy mechanism for 3+ cation incorporation. At higher Al contents (>0.08 pfu), the relationship between Al and Fe^{3+} concentrations becomes stronger and a 1:1 interdependence is approached. This results from the emerging dominance of a coupled substitution mechanism. The data in Figure 8 have been parameterized with a simple empirical relationship to describe the Fe^{3+} content as a function of bulk Fe and Al:

$$x_{\text{Fe}^{3+}} = \alpha_0 x_{\text{Al}} + \alpha_1 x_{\text{Al}}^2 + \beta_0 x_{\Sigma\text{Fe}}. \quad (7)$$

The fitted parameters are $\alpha_0 = -0.13 \pm 0.07$, $\alpha_1 = 3.1 \pm 0.5$, and $\beta_0 = 0.39 \pm 0.03$. Using this empirical relationship, both Mg-Pv Fe^{3+} contents obtained by our calculations and direct measurements are reproduced within ± 0.012 pfu and ± 0.022 pfu for 68% and 95% confidence limits, respectively, for the 78 data points plotted in Figure 8.

[34] Although it may be possible to produce a perovskite with $\text{Fe}^{3+}/\Sigma\text{Fe} = 1$, for any given bulk Fe content a limit in Al content will be reached where insufficient Fe^{3+} is available on the A-site to balance Al in the B-site. Experiments employing mid-oceanic ridge basalt (MORB) bulk compositions indicate that it is indeed possible to produce Al-rich Mg-Pv where insufficient Fe^{3+} exists to balance Al in the B-site [*Litasov et al.*, 2003]. The stoichiometry of such samples, shown in Figure 9, is such that Al must start to also substitute onto the A-site, and Fe^{3+} contents decrease as a result of charge balance on the B-site being now being maintained partly by Al on the A-site. This most likely occurs because at any given pressure there is a limit to the solubility of Al on the B-site, which is in accordance with experiments performed in the Fe-free system [*Irifune et al.*, 2002]. It then becomes energetically more favorable for Fe^{3+} to be replaced by Al, and the dependence of Al-bearing Mg-Pv on Fe^{3+} decreases. This is shown by the downward trending bulk Fe contours in Figure 9, which indicates a 1:1 inverse dependence of Fe^{3+} on Al. It can be seen that the high Al and Fe Mg-Pv samples synthesized by *Litasov et al.* [2003] fit these downward trending contours very well. However, it is unclear whether or not the trend can be applied for the high Al but low Fe compositional range as no data exist for these compositions.

4.4. Implications for Fe^{3+} Concentrations in the Lower Mantle

[35] Knowledge of the chemical composition of Mg-Pv as a function of bulk rock composition, pressure and temperature is important for most geophysical calculations of lower mantle properties. The currently prevailing view is that the lower mantle dominantly has a peridotitic or bulk silicate Earth (BSE) composition [e.g., *Irifune et al.*, 2010]. However, it must also contain chemical heterogeneities arising from the subduction of oceanic lithosphere with crustal and depleted mantle lithologies that were originally basaltic and harzburgitic. Because Al and bulk Fe contents of these

rocks are quite different, Mg-Pv that forms in these different lithologies should contain different concentrations of these two elements. In addition, the Al contents of Mg-Pv in some bulk compositions will change due to the disappearance or appearance of aluminous phases such as majoritic garnet, depending on phase stability. Using the relationship among Fe^{3+} , Al^{3+} and bulk Fe of Mg-Pv obtained in this study, it is possible to determine the range in Fe^{3+} contents for plausible lower mantle rock compositions (Figure 9).

[36] At the very top of the lower mantle, ~ 660 km depth, the Mg-Pv that initially forms by the breakdown of ringwoodite in a peridotitic i.e., BSE bulk composition [e.g., *McDonough and Sun*, 1995] will have a relatively low Al content of 0.02 pfu because Al will reside mainly in coexisting majoritic garnet [*Wood*, 2000], which persists within the top 30–40 km of the lower mantle. Some uncertainty exists as to whether Mg-Pv Fe^{3+} contents depend on f_{O_2} at these low Al concentrations. Analyses of Al-free Mg-Pv synthesized in Fe capsules have reported $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratios < 0.1 – 0.2 [*Lauterbach et al.*, 2000]. The total Fe content of Mg-Pv at the top of the lower mantle will be approximately 0.09 pfu and the corresponding Fe^{3+} content will, therefore, be of the order of 0.01–0.02. At these low Al contents it is likely that a significant proportion of the 3+ cations substituting into the Mg-Pv Si site (B-site) will have charge balance maintained by the creation of oxygen vacancies. At greater depths, however, majoritic garnet will break down and by 720 km depth the bulk of the Al in a BSE composition will partition into Mg-Pv, resulting in an Al content in the range 0.08–0.11 [*Irfune*, 1994]. For a BSE composition, Mg-Pv with this level of Al will persist throughout the lower mantle [*Irfune et al.*, 2010], resulting in an Fe^{3+} content of 0.05–0.06 pfu and $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratios in the range 0.5–0.6. At such concentrations it is likely that the coupled substitution of Fe^{3+} and Al starts to dominate as the main substitution mechanism. Similar $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratios (0.52–0.67) were measured by *Irfune et al.* [2010] for Al-bearing Mg-Pv synthesized up to pressures of 47 GPa. The bulk Fe content of Mg-Pv in a BSE composition at depths greater than 720 km should be approximately 0.10 pfu. Although *Irfune et al.* [2010] reported a drop in $K_D(\text{app})$ for Mg-Pv and Fp formed from a BSE composition between 25 and 47 GPa, this drop corresponds to a change in bulk Fe content of Mg-Pv of only 0.02 pfu, which is relatively small and is unlikely to significantly influence mineral properties. It is, however, hard to explain this drop, which seems to result mainly from Fe^{2+} -Mg exchange. Over the same pressure interval the model derived in this study, from Al-free DAC results described above, would predict a drop of only 0.006 pfu.

[37] Subducted depleted lithospheric mantle with a harzburgitic bulk composition [e.g., *Michael and Bonatti*, 1985] will form Mg-Pv with a similar Al content to that formed from a BSE composition at the top of the lower mantle ($< \sim 0.02$ pfu) [*Irfune and Ringwood*, 1987]. The low Al content, which is inherent to the bulk composition, will persist, however, throughout the Mg-Pv stability field. Given bulk Fe contents of 0.09 pfu for Mg-Pv in a harzburgite composition, the Fe^{3+} content will lie, as described above, in the range 0.01–0.03 pfu and the oxygen vacancy substitution mechanism will likely remain dominant.

[38] A number of experiments have been performed to determine the mineralogy of a mid-oceanic ridge basalt (MORB) composition in the lower mantle. At 25–30 GPa, Mg-Pv containing Al in the range 0.25–0.35 pfu and bulk Fe between 0.32 and 0.40 pfu has been reported [*Hirose et al.*, 1999; *Litasov and Ohtani*, 2005; *Ono et al.*, 2001] with the variation mainly reflecting the particular bulk composition selected. As described at the end of the previous section, the measured $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratios for such Al-rich perovskites are about 0.6 [*Litasov et al.*, 2003], with Fe^{3+} contents on the order of 0.20–0.25 pfu that are insufficient to allow all Al to be charge balanced on the Si site. Some Al is therefore accommodated in the Mg-site, reducing Fe^{3+} and causing the negative Fe^{3+} dependence on Al shown in Figure 9. The Mg-Pv Al contents of MORB are in the range where this AlAlO_3 substitution starts to take over from the $\text{Fe}^{3+}\text{AlO}_3$ substitution. The situation at pressures higher than 25–30 GPa is more uncertain. Mg-Pv may be able to accommodate more Al on the Si site at higher pressures, as suggested by the study of *Nishio-Hamane et al.* [2005], who synthesized perovskite with 25% of the $\text{Fe}^{3+}\text{AlO}_3$ component at 50 GPa.

[39] Variations in rock composition in the lower mantle will, therefore, result in large variations in Mg-Pv Fe^{3+} contents due to differences in Al and bulk Fe contents. These changes also reflect differences in the dominant 3+ cation substitution mechanisms, with the corresponding Mg-Pv 3+ cation components varying from $\text{MgAlO}_{2.5}$ at low Al contents to $\text{Fe}^{3+}\text{AlO}_3$ for BSE compositions and additionally AlAlO_3 for MORB bulk compositions. Most importantly, however, such changes in the 3+ substitution mechanism in Mg-Pv and the corresponding variations in Fe^{3+} and oxygen vacancy concentrations probably result in differences in physical properties between major chemical heterogeneities in the mantle, including viscosity, electrical conductivity, thermal conductivity and elasticity.

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