

# Week 2-8

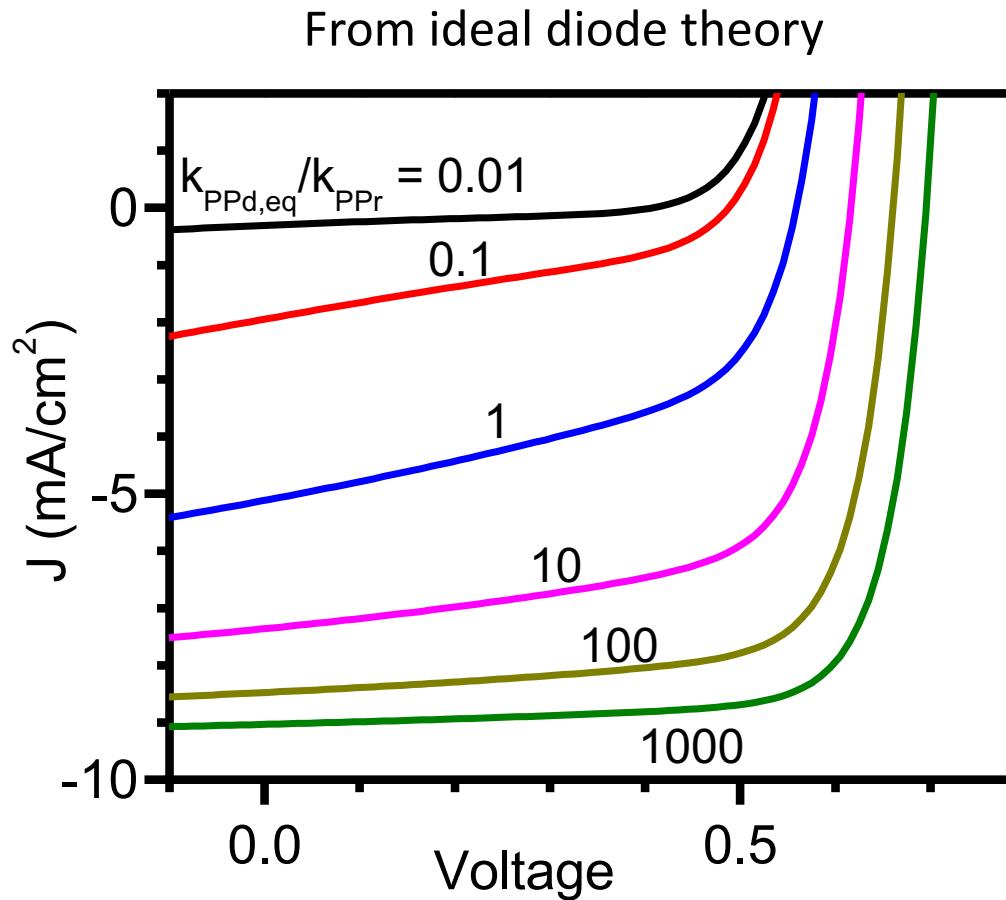
## Optical Detectors 3

Morphology  
Materials  
Active Region Architecture  
Transparency

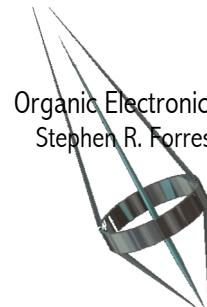
Chapter 7.4

Organic Electronics  
Stephen R. Forrest

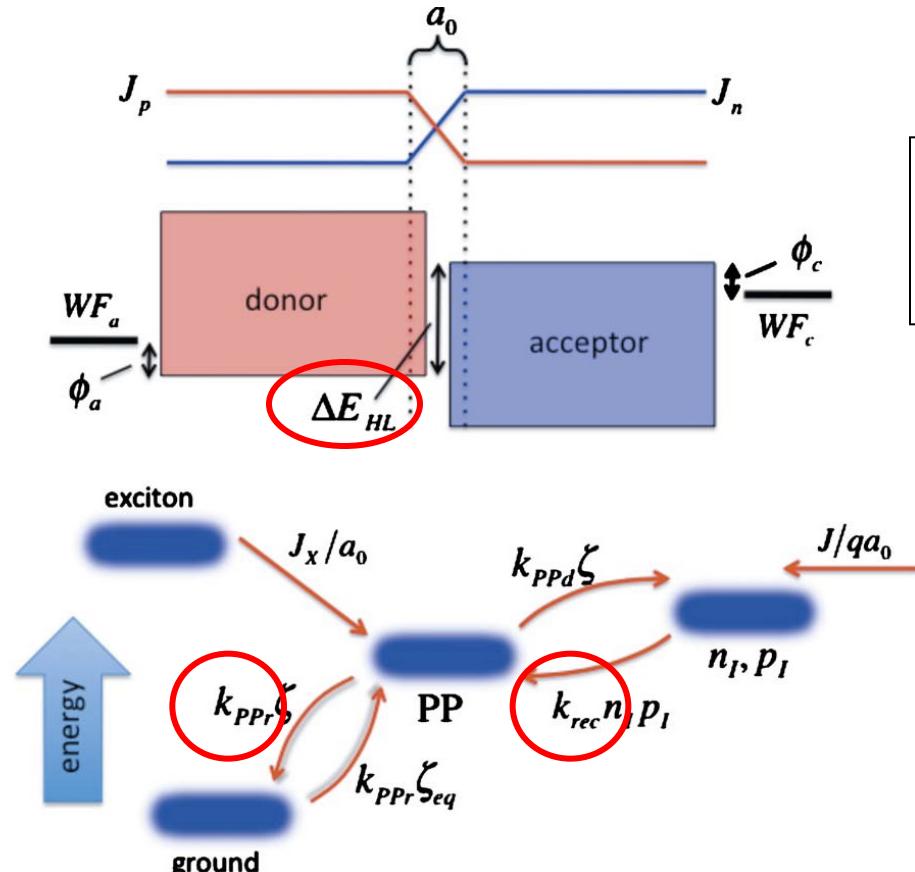
# The central importance of morphology



- PP recombination  $\Rightarrow$  Reverse Slope
- Best morphologies limit  $k_{PPr}$  at interface:
  - **Steric hindrance**
  - **Disorder at interfaces/order in the bulk**



# Morphology Dependence of $V_{OC}$



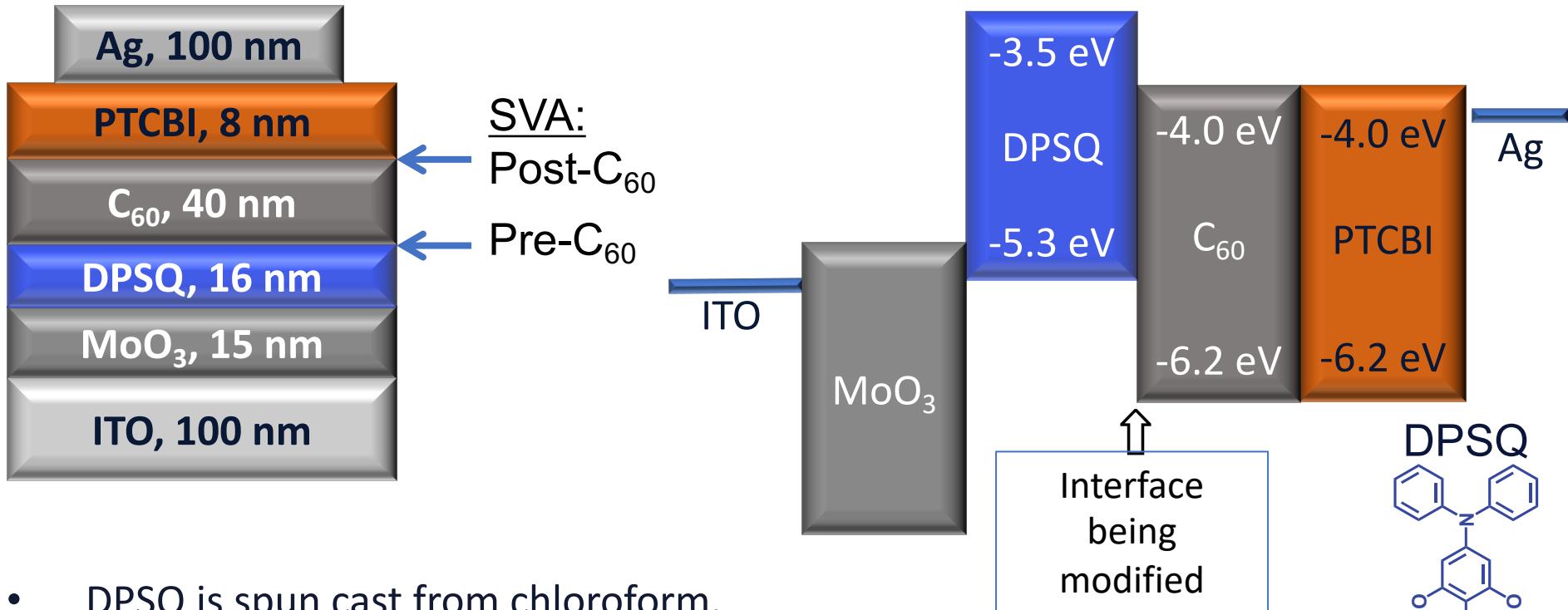
$$qV_{OC} = \Delta E_{HL} - nk_B T \ln \left[ \frac{k_{PPr}}{k_{PPd}} \frac{k_{rec} N_L N_H}{J_X / \alpha_0} \right] *$$

- Material choice determines:
  - $\Delta E_{HL}$  (HOMO-LUMO Gap)
  - Steric hindrance (MO overlap)
- Device processing/morphology can limit  $V_{OC}$  losses:
  - $k_{rec}$  (PP formation)
  - $k_{PPr}$  (PP recombination)

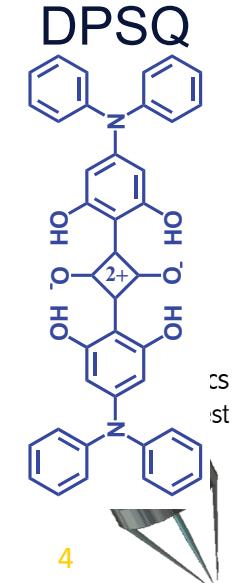
\*Giebink *et al.* Phys. Rev. B 82 155305 2010

# Controlling Interface Morphology to Affect Recombination ( $\Rightarrow V_{OC}$ )

SVA of Squaraine/Fullerene Solar Cells (see Ch. 4.7)

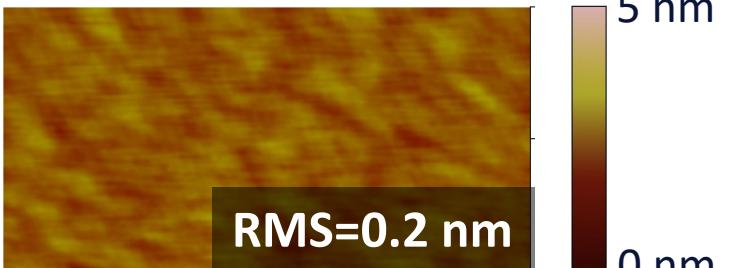


- DPSQ is spun cast from chloroform.
- Other layers deposited by thermal evaporation.
- **Vary D-A interface order via solvent vapor annealing (SVA):**
  - 10 min exposure to dichloromethane vapor to “anneal” squaraine component.



# SVA Pre- $C_{60}$

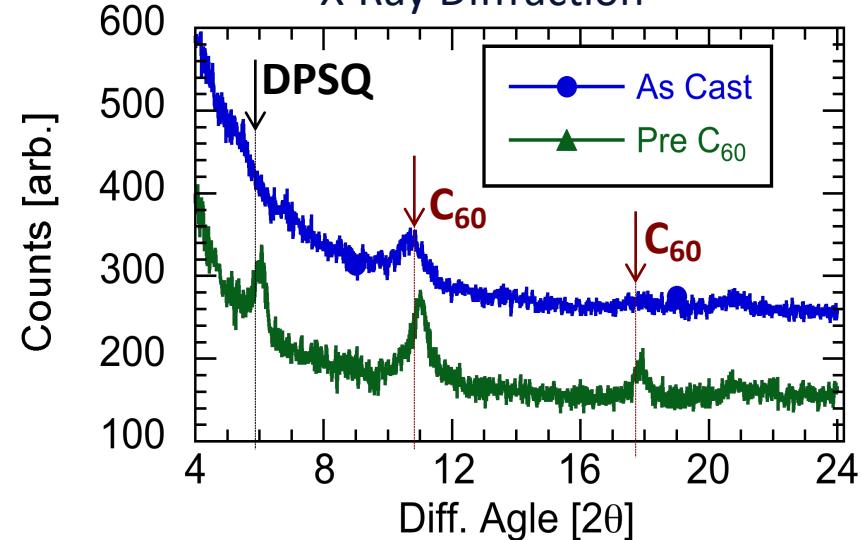
As Cast DPSQ



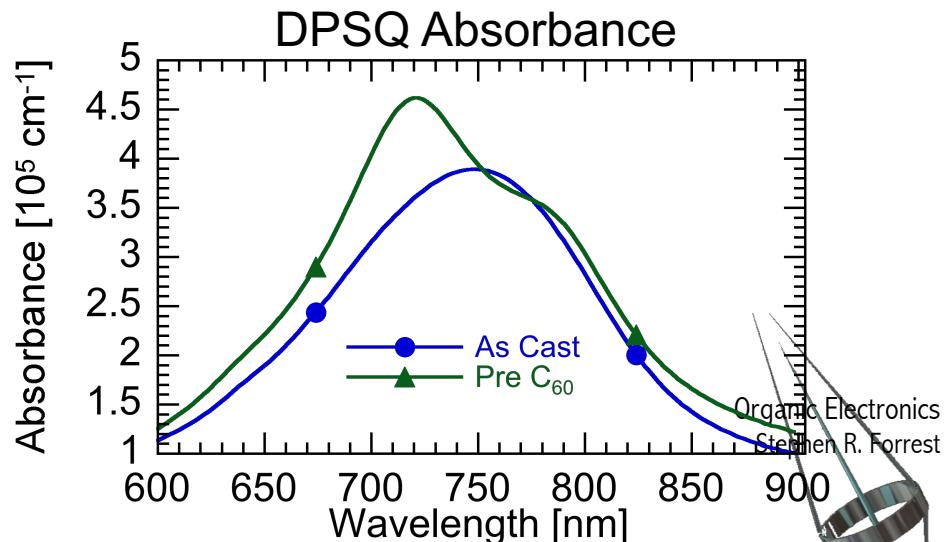
DPSQ SVA pre- $C_{60}$



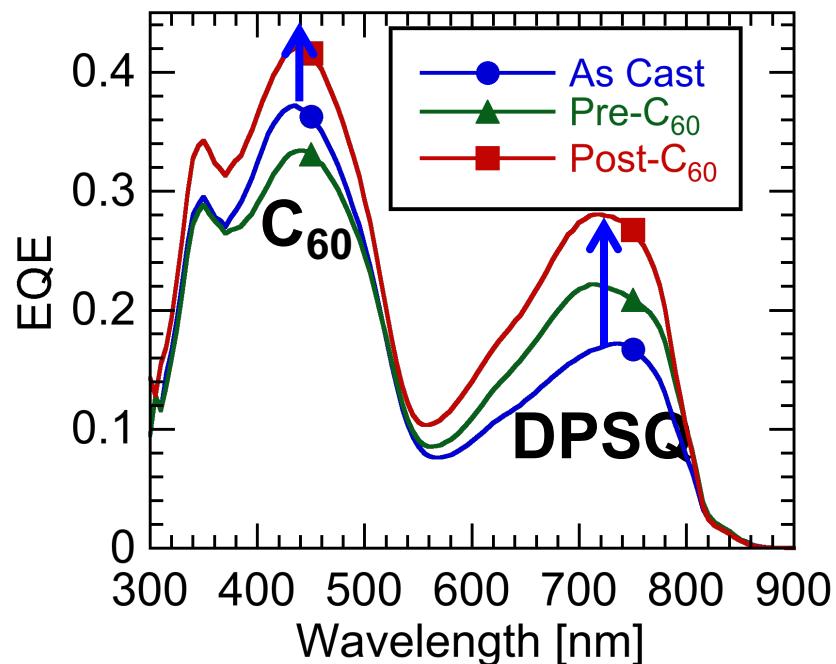
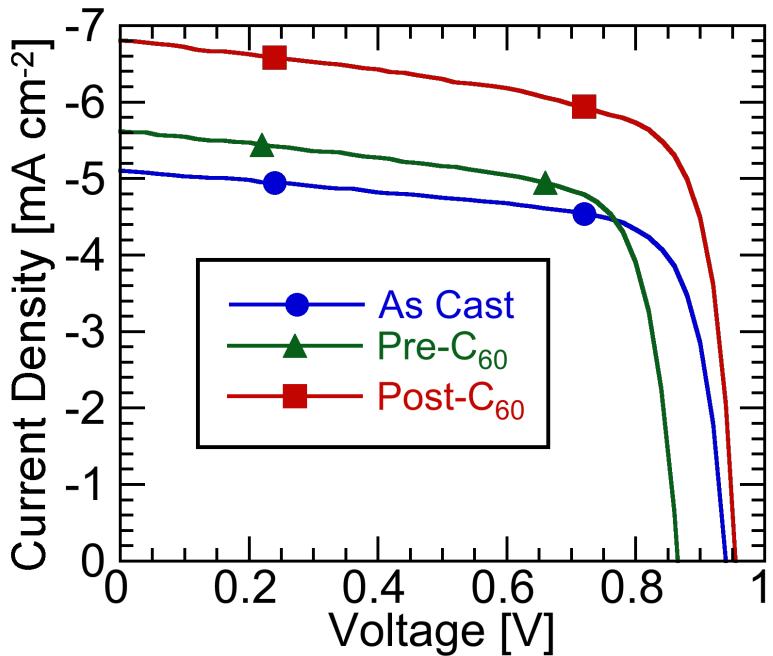
X-Ray Diffraction



- SVA Pre  $C_{60}$ :
  - Crystallizes and roughens DPSQ.
  - Crystalline DPSQ templates *quasi-epitaxial*  $C_{60}$  growth.



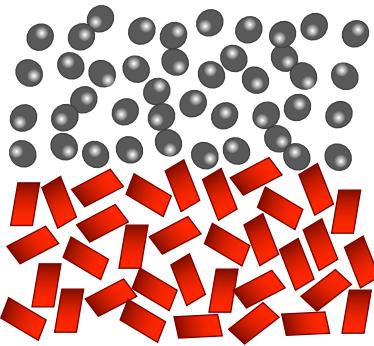
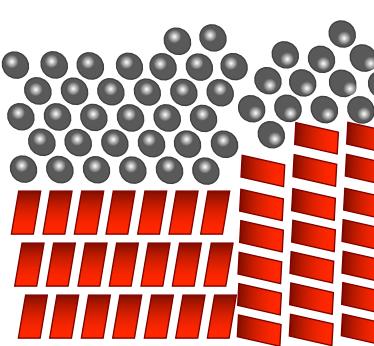
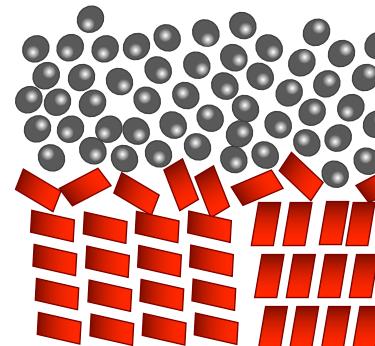
# Devices with SVA Post- $C_{60}$

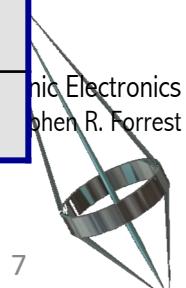


- SVA post- $C_{60}$ 
  - DPSQ EQE  $\uparrow$  80%.
  - $J_{SC} \uparrow$  25%.
- No loss in  $V_{OC}$ 
  - $k_{PPr}$  unchanged.

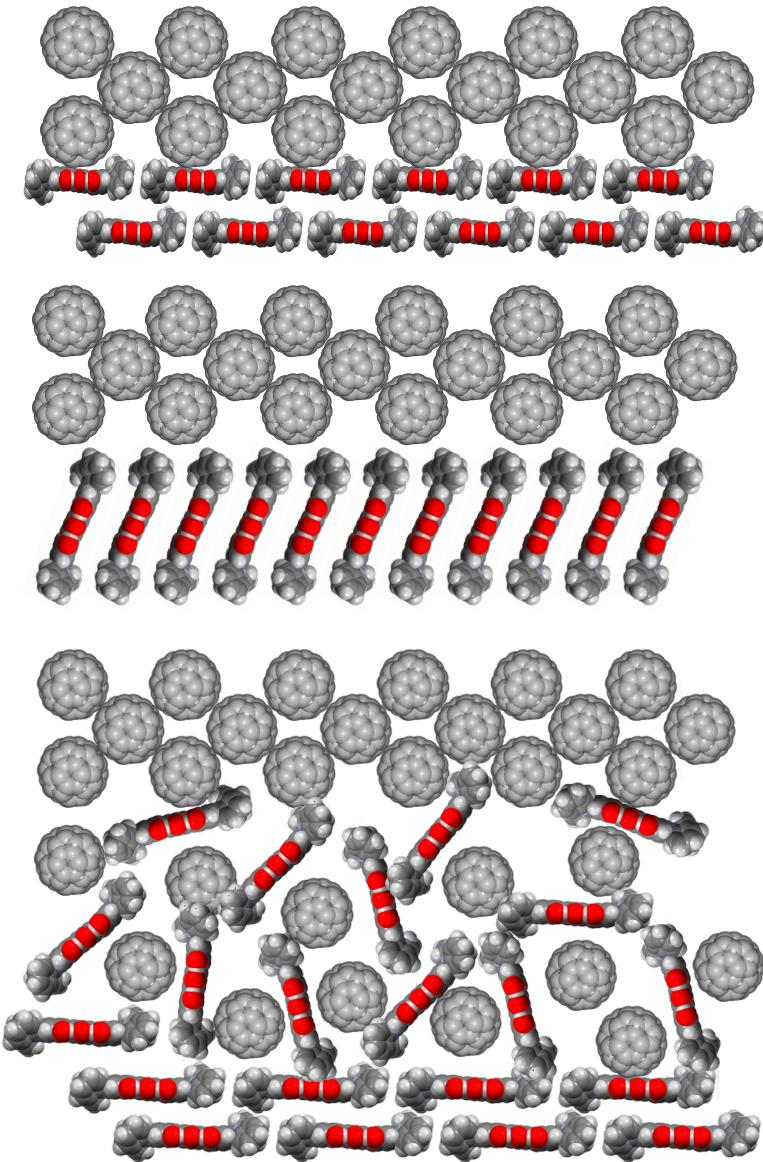
Process	$J_{SC}$ [mA cm <sup>-2</sup> ]	$V_{OC}$ [V]	FF [%]	$\eta_P$ [%]
As Cast	$5.3 \pm 0.3$	0.94	73	$3.6 \pm 0.2$
Pre- $C_{60}$	$5.6 \pm 0.3$	0.86	70	$3.4 \pm 0.2$
Post- $C_{60}$	$7.0 \pm 0.4$	0.96	71	$4.8 \pm 0.3$

# Achieving the Ideal Interface Morphology

$C_{60}$			
	As Cast	Pre $C_{60}$	Post $C_{60}$
Bulk DPSQ	Amorphous	Ordered	Mod. Order
Bulk $C_{60}$	Weak order	Ordered	Weak Order
Interface	Disordered	Ordered	Disordered
Surface	Smooth	Rough	Smooth
$k_{PPr}$	Low	High	Low
$V_{OC}$	High	Low	High
$J_{SC}$	Low	Moderate	High



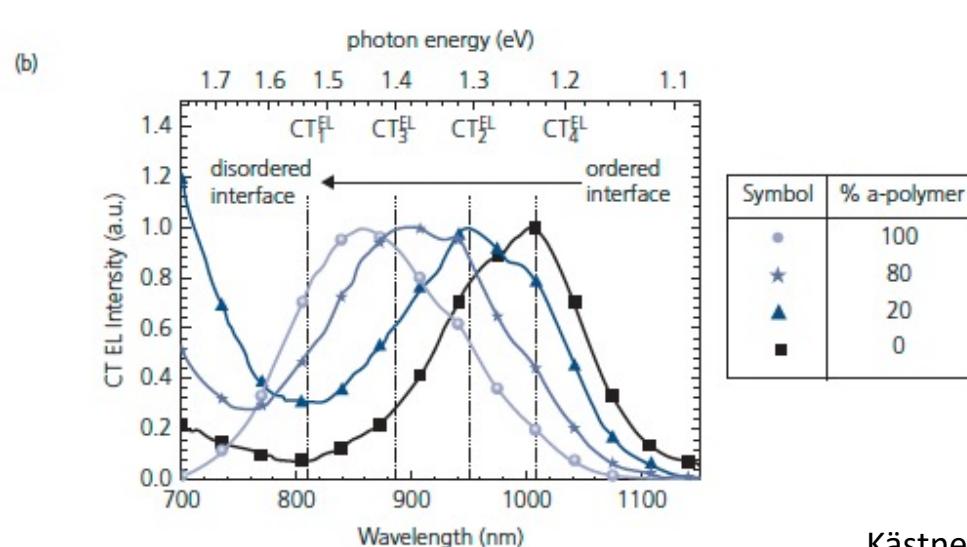
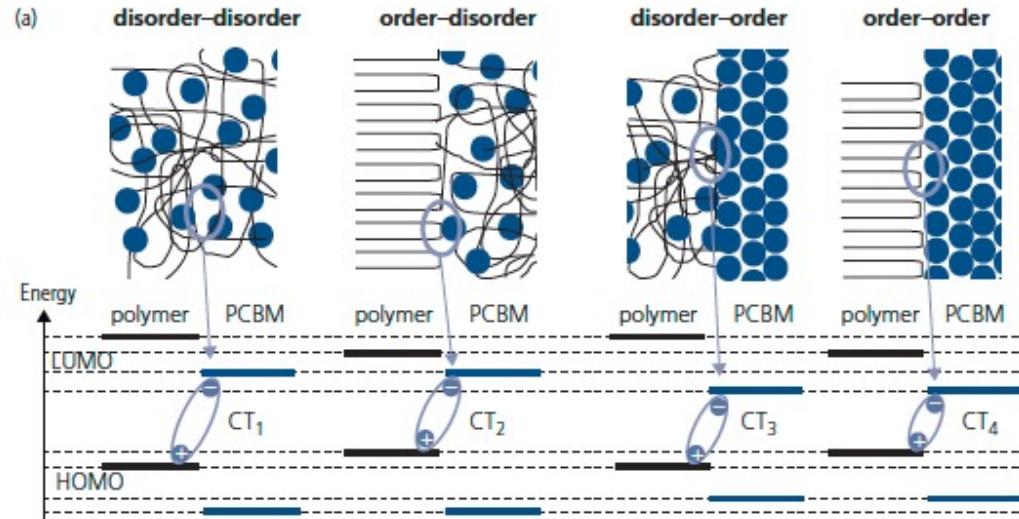
# Morphology vs. $V_{OC}$



$$qV_{OC} = \Delta E_{HL} - nk_B T \ln \left[ \frac{k_{PPr}}{k_{PPd}} \frac{k_{rec} N_L N_H}{J_X / \alpha_0} \right]$$
$$k_{rec} = \gamma = \frac{q}{\varepsilon} (\mu_e + \mu_h)$$

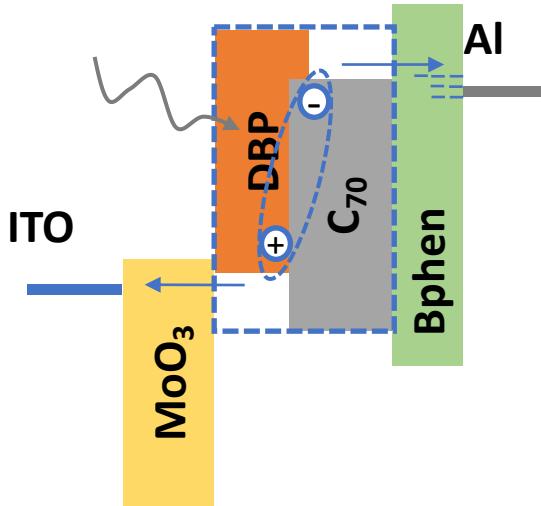
- Worst case scenario: perfectly ordered crystalline interface and bulk, Face-on .
  - High  $k_{PPr}$  and  $k_{rec}$
- Better Scenario I: Perfectly crystalline and end-on orientation
- Even Better Scenario II: crystalline bulk, intermixed interface
  - Poor coupling between like-molecules (C<sub>60</sub>-C<sub>60</sub> and SQ-SQ) reduces PP formation ( $k_{rec}$ ) probability.
  - Overcomes enhanced  $k_{PPr}$  due to facial contact

# Disorder Dependent CT State Energy at Polymer Junctions

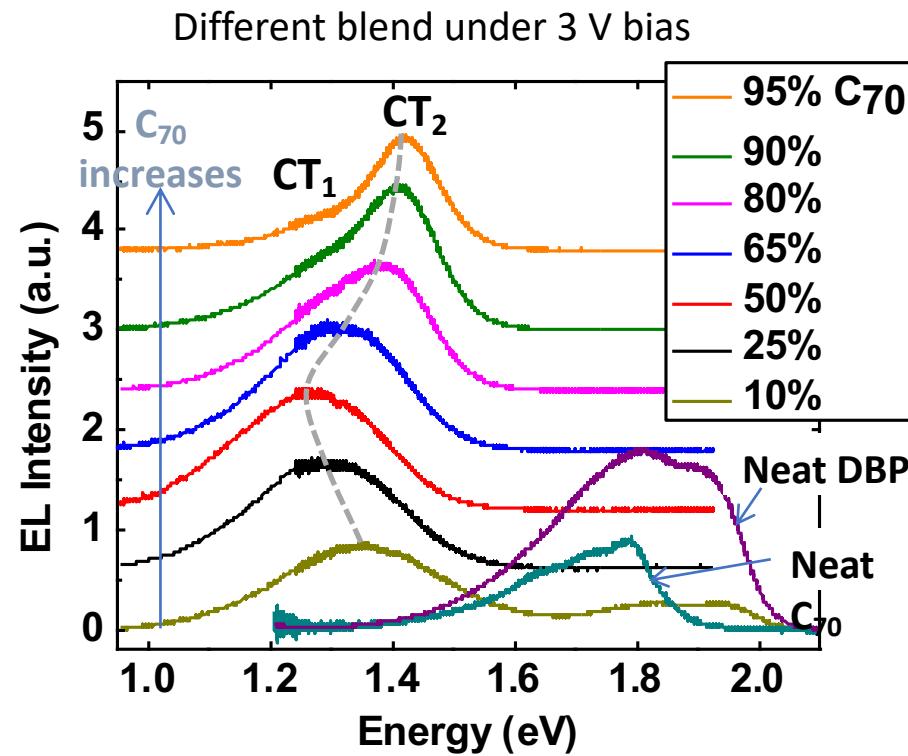


- Morphology at functionalized PPV:PCBM HJs controlled by functionalization and blends
- Qualitatively similar phenomena as for small molecule (DPSQ:C60) junctions

# Morphology Controls CT State Energy

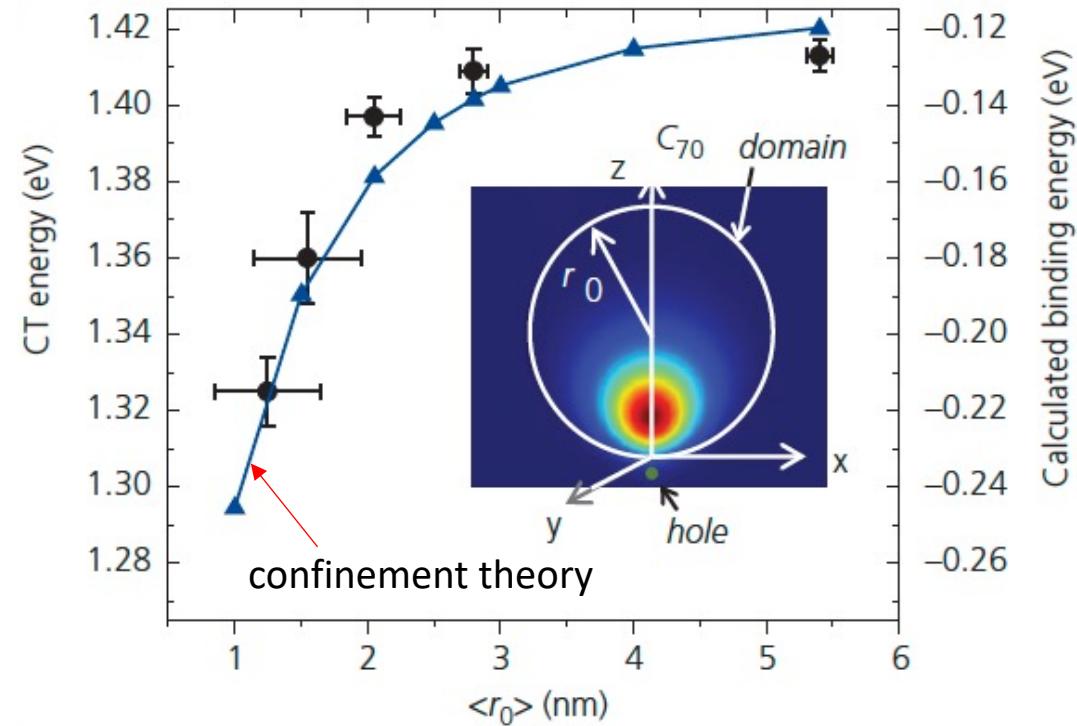
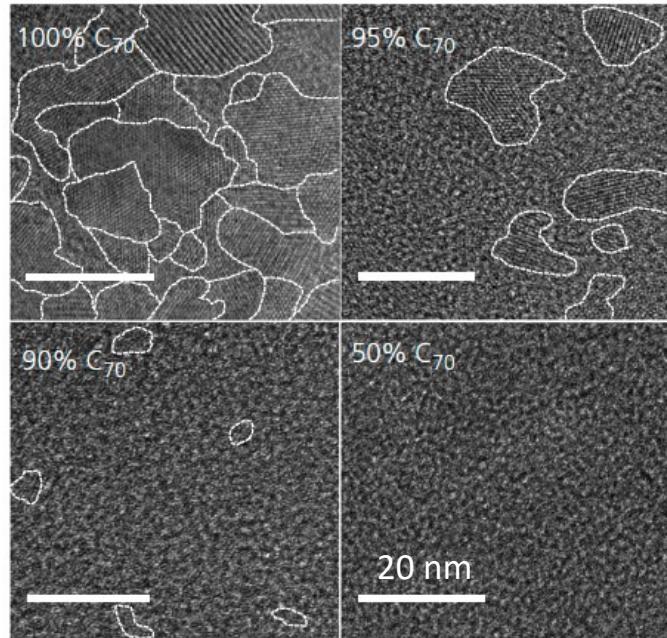


CT states are directly excited optically, or by injection from contacts (operating the OPV in the OLED mode).



Shift in energy of CT<sub>2</sub> due to morphology changes as the amount of C<sub>70</sub> in DBP changes

# CT Energy Increases with C<sub>70</sub> Domain Size ⇒ Exciton Confinement



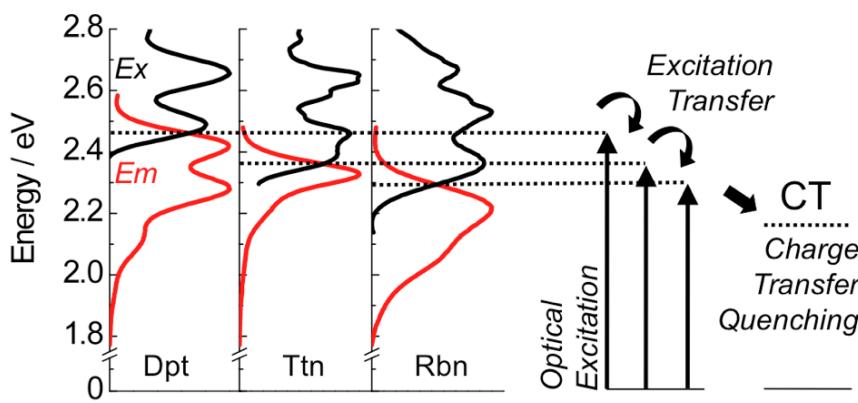
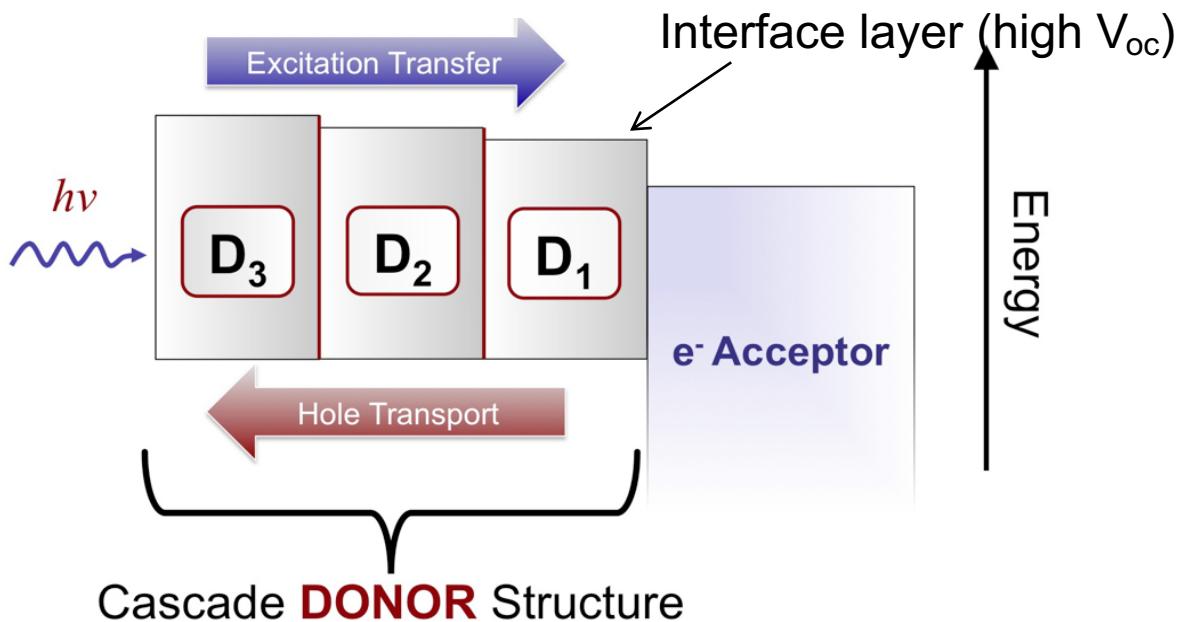
$$H = -\frac{\hbar^2}{2m_e^*} \nabla^2 + V(\mathbf{r}),$$

$$V(\mathbf{r}) = \begin{cases} -\frac{q^2}{4\pi\epsilon_0\epsilon_{r,C_{70}}} |\mathbf{r}-\mathbf{a}| & (|\mathbf{r}-\mathbf{r}_0| < r_0) \\ -\frac{q^2}{4\pi\epsilon_0\epsilon_{r,DBP}} |\mathbf{r}-\mathbf{a}| + \phi_b & (|\mathbf{r}-\mathbf{r}_0| > r_0) \end{cases}$$

- Electron confined to C<sub>70</sub> domain, hole immobilized on DBP
- Barrier ( $\phi_b$ ) exists at C<sub>70</sub> domain periphery

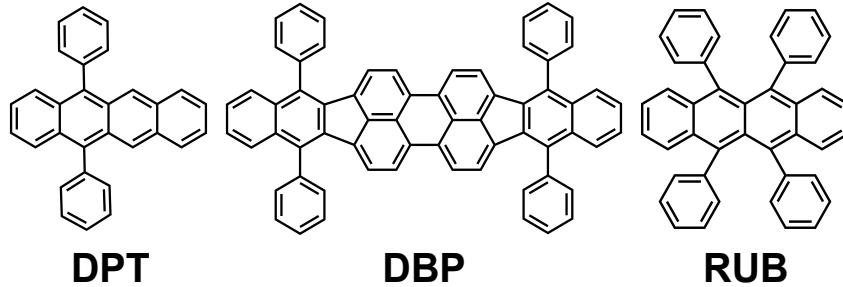
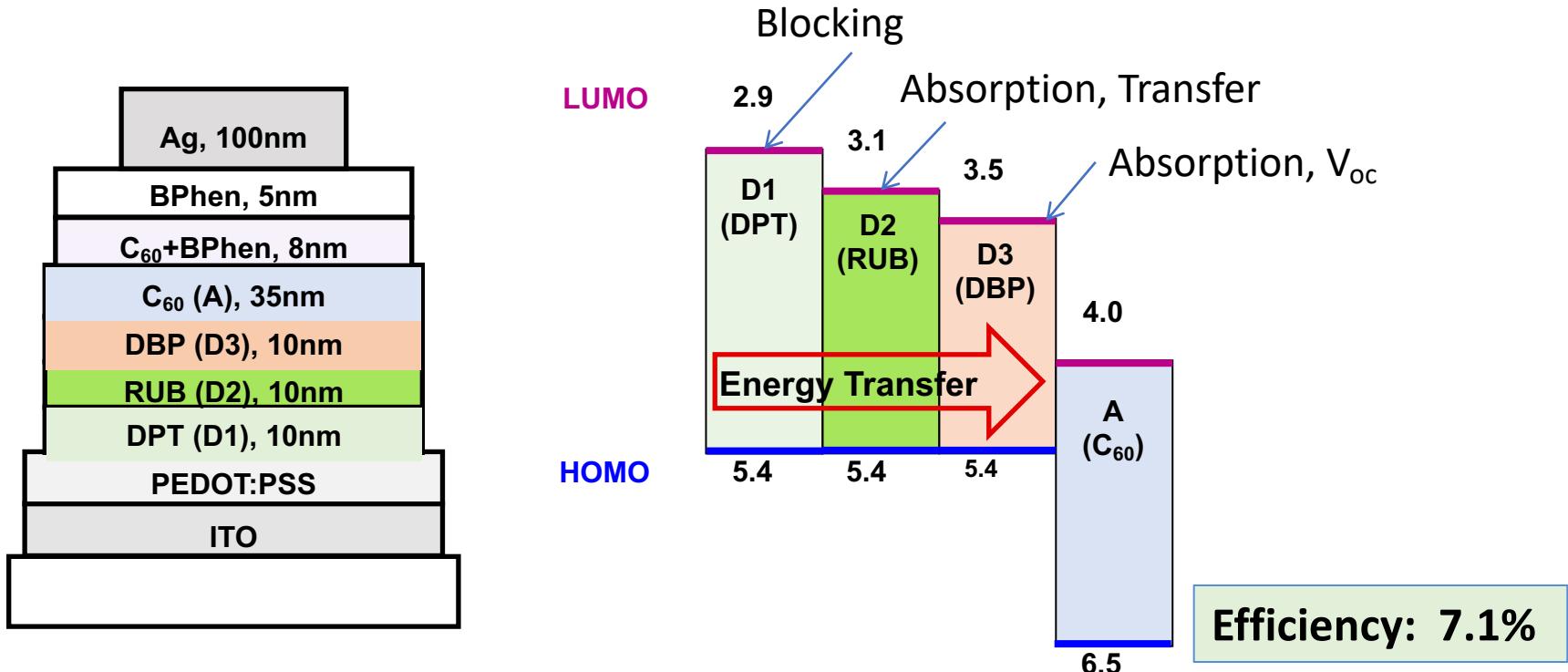
# New Architectures for Increased Spectral Coverage

## Exciton Cascades



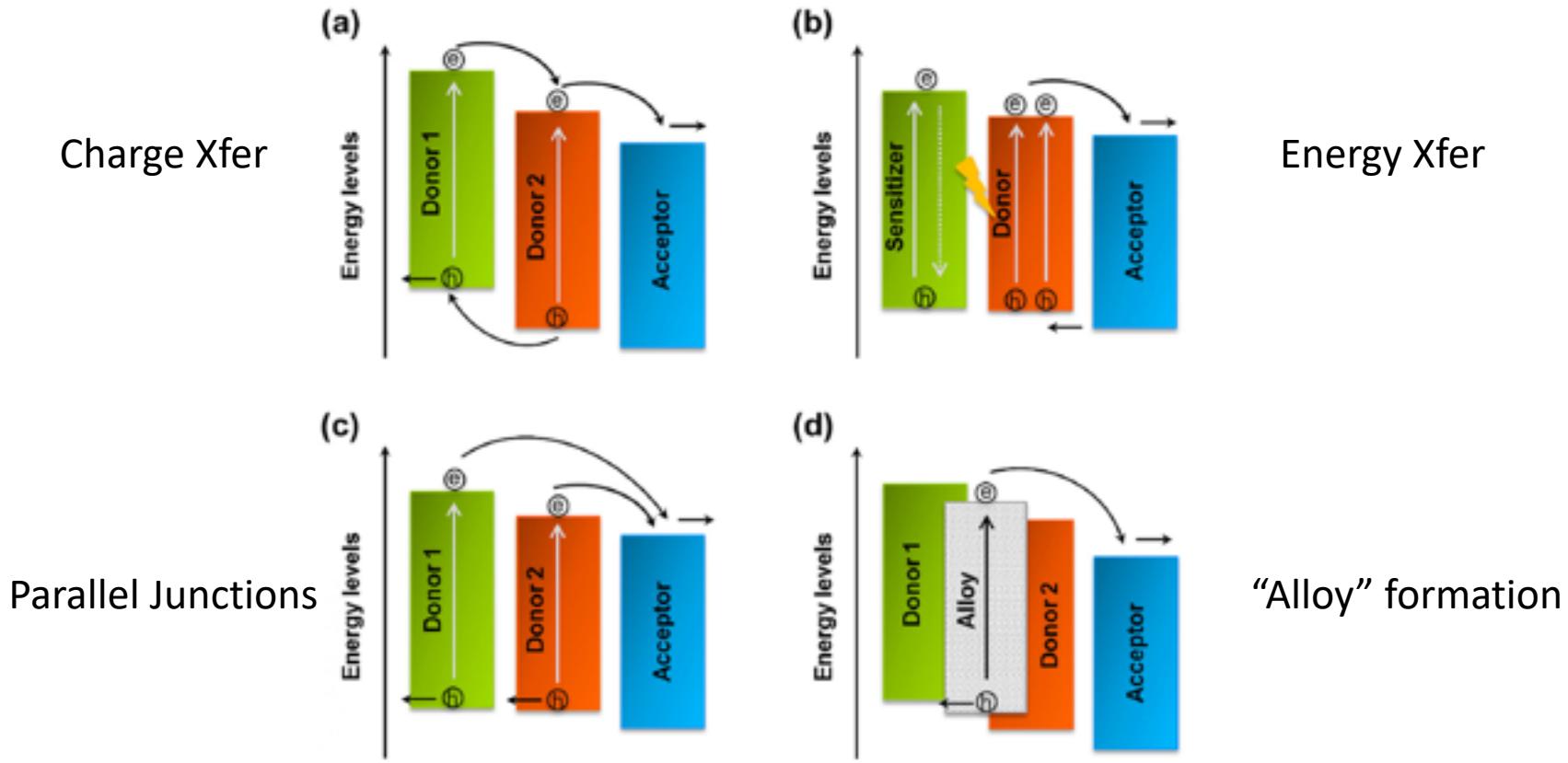
Schlenker, et al., *Chem. Mater.*, **23**, 4132 (2011).

# High Efficiency Donor Cascade Device



O. L. Griffith & S. R. Forrest, *Nano Letters*, 14, 2353 (2014).

# Ternary BHJs Increase Solar Coverage

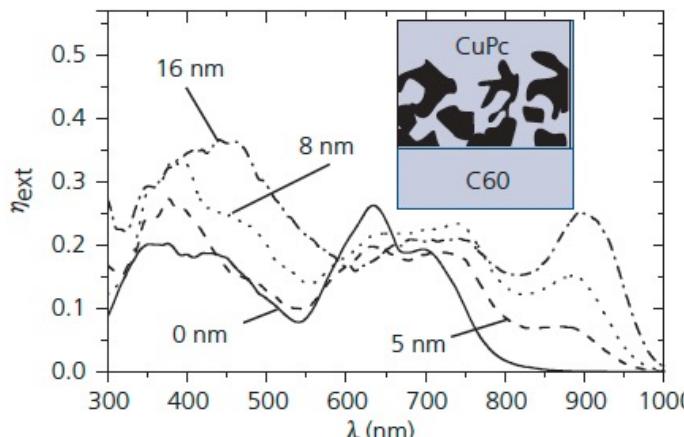
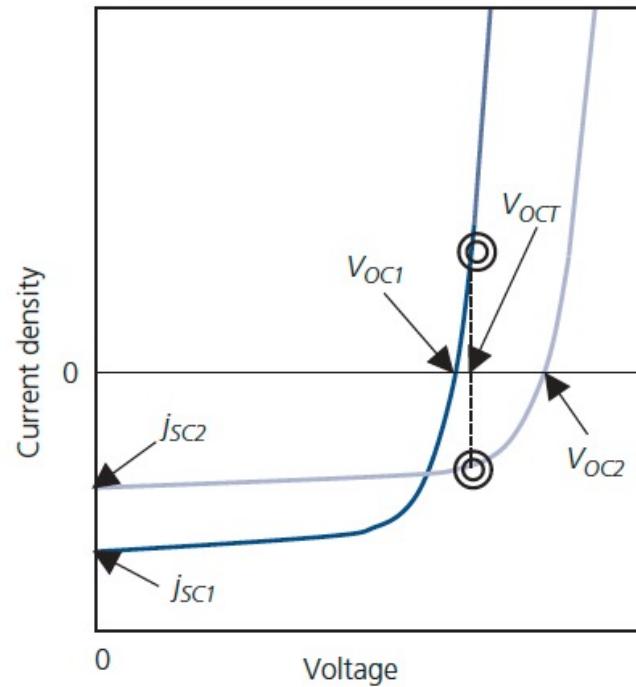


## Features of ternary blends:

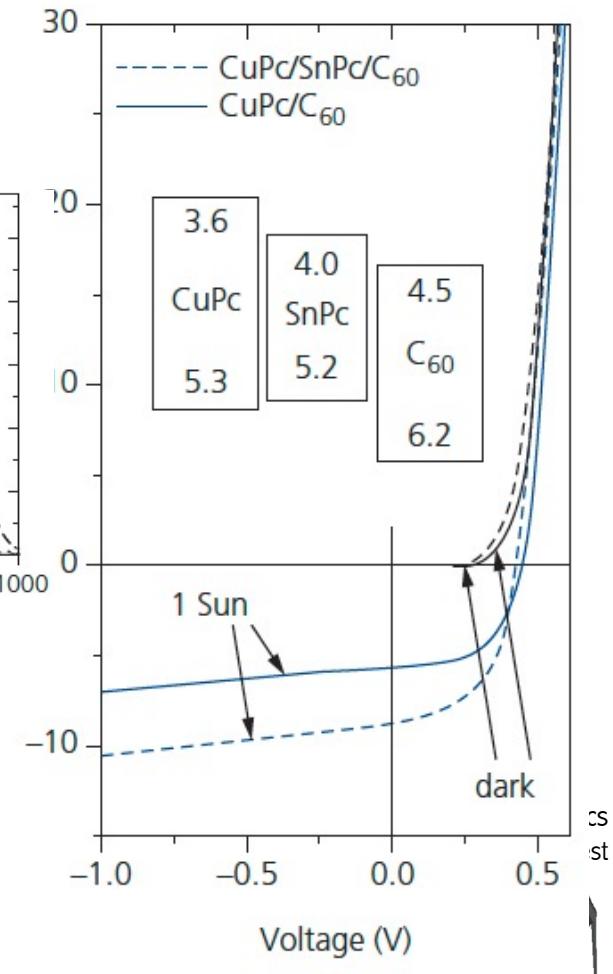
- $V_{OC}$  of the ternary lies between the extremes of the two subcell junctions
- Materials chosen to cover solar spectrum
- Can be DA<sub>1</sub>A<sub>2</sub> or AD<sub>1</sub>D<sub>2</sub> junctions
- Morphology is key
- Probably more than one process governs performance
- Molecular alloy formation unlikely

# Parallel Junction Ternary OPVs

- $V_{OC}$  intermediate between individual binary OPVs



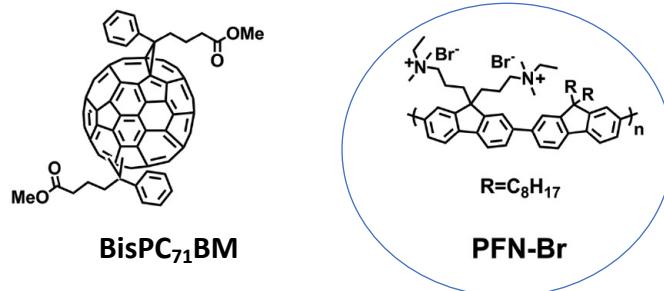
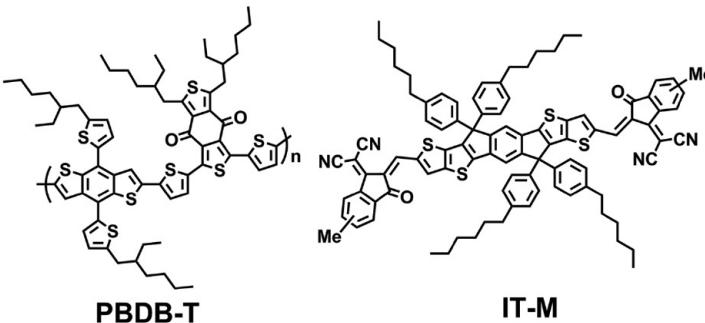
Increased spectral coverage



Increased efficiency



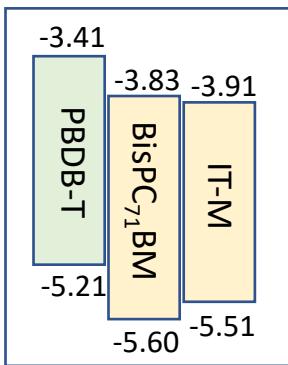
# Example DA<sub>1</sub>A<sub>2</sub> ternary BHJ



D:A <sub>1</sub> :A <sub>2</sub> ratio	$V_{OC}$ (V)	$j_{SC}^{(a)}$ (mA/cm <sup>2</sup> )	FF	$\eta_P$ (max) (%)	$\eta_P$ (ave) <sup>(b)</sup> (%)
1:1:0	0.937	16.7	0.69	10.80	10.45
1:1:0.2	0.952	17.4	0.74	12.20	11.75
1:0:1	1.02	10.6	0.58	6.25	5.86

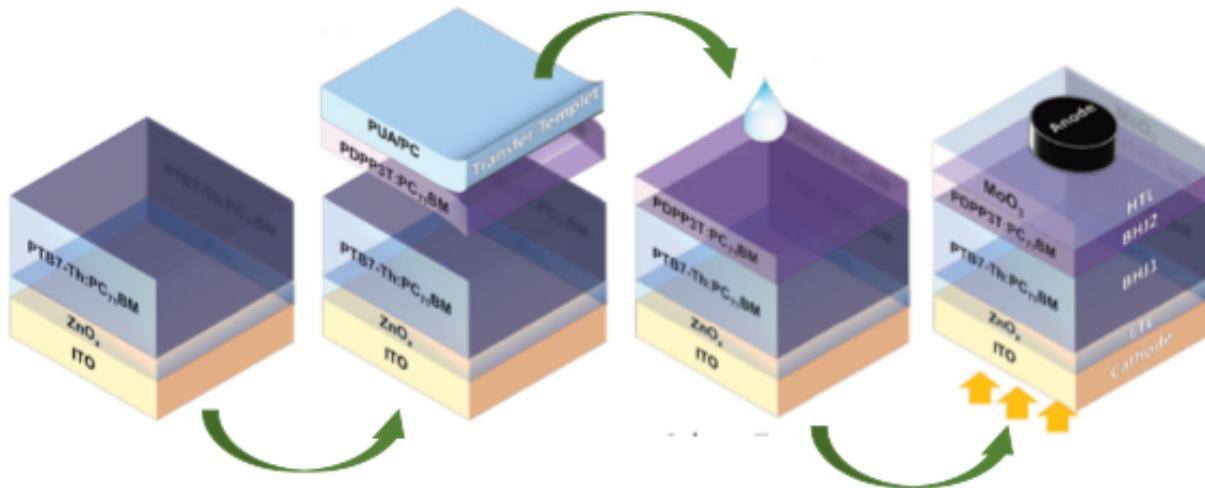
- Cell area = 4 mm<sup>2</sup>.
- Sample size = 100 diodes

Exciton blocker



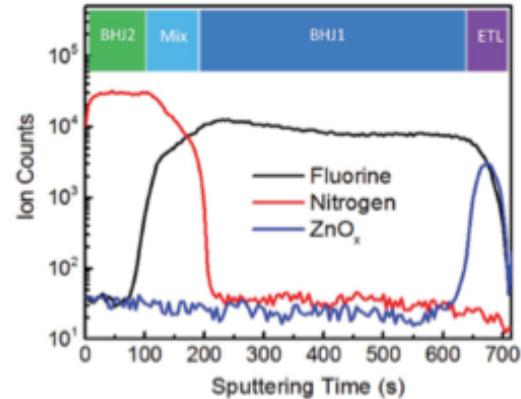
W. Zhao, et al., *Advanced Materials*, **29**, 2, 2017.

# Solution-Processed Bi-tertiary OPV

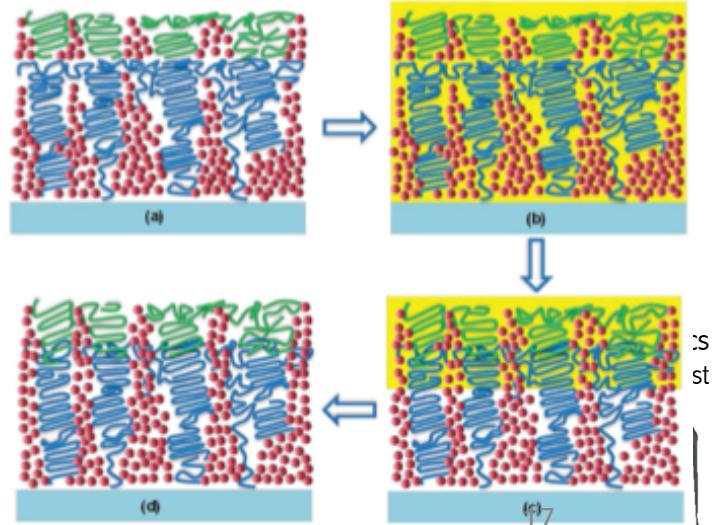


Intermixing of two donors creates continuous hole and electron conduction

Cell <sup>(a)</sup>	$V_{OC}$ (V)	$j_{SC}^{(b)}$ (mA/cm <sup>2</sup> )	FF	$\eta_P$ (max) (%)	$\eta_P$ (ave) <sup>(c)</sup> (%)
BHJ1	0.81	18.5	0.70	10.5	10.3
BHJ2	0.69	7.4	0.71	3.6	3.5
BHJ1/2	0.77	23.8	0.67	12.3	11.9

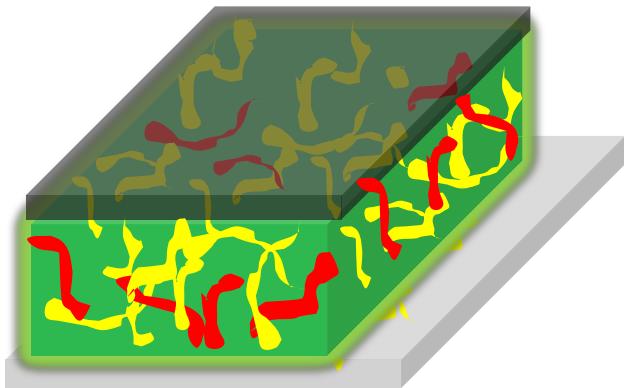


(b)

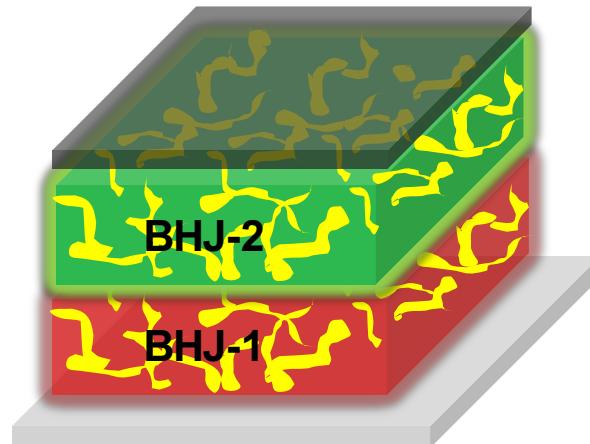


# Series Connecting Two BHJs: Bi-Ternary OPVs

(a)

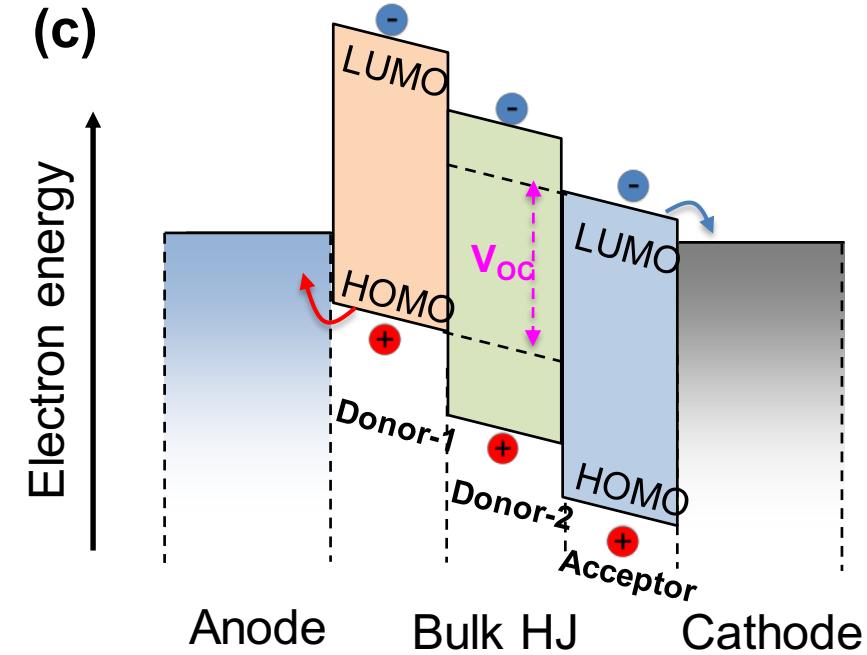


(b)

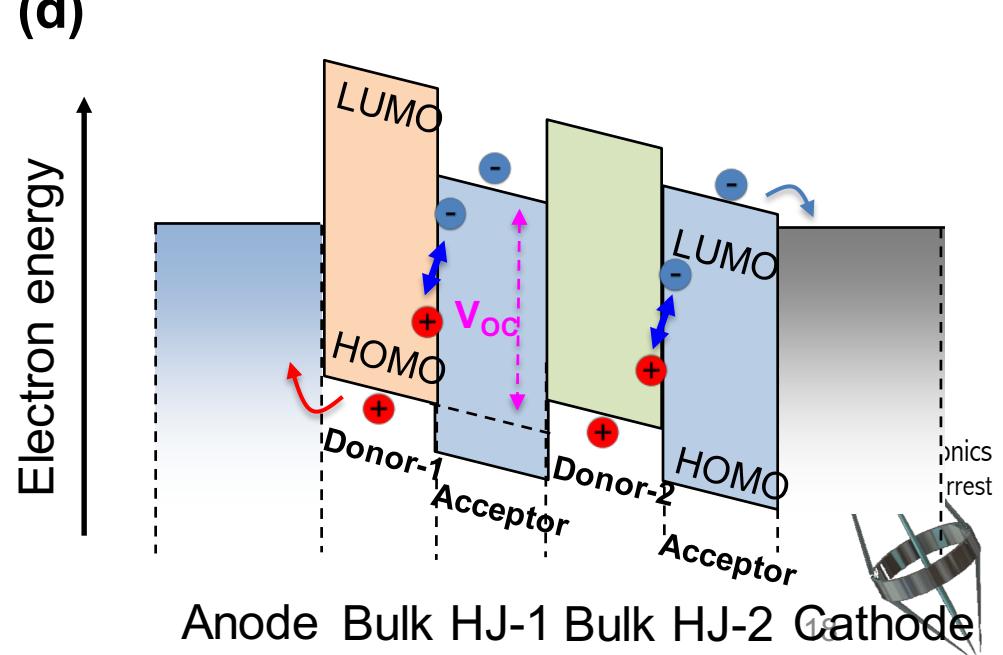


Classic Ternary Solar Cells

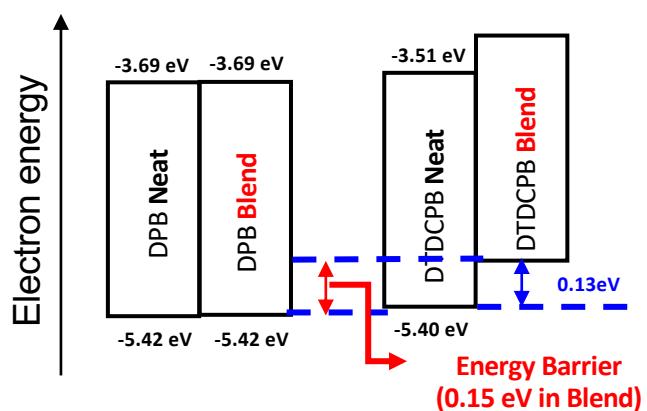
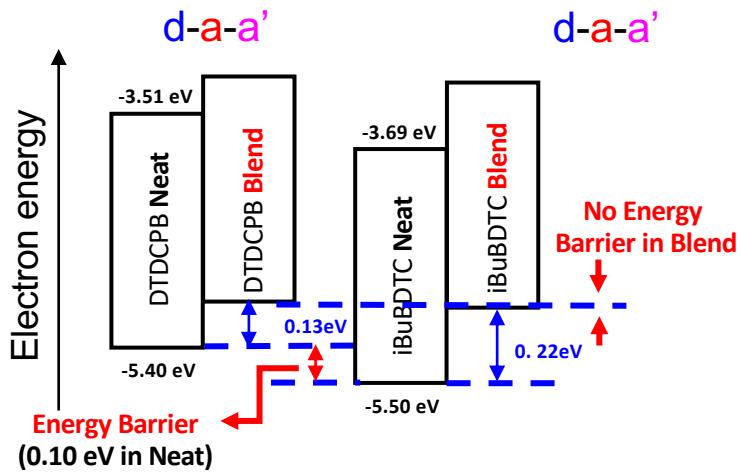
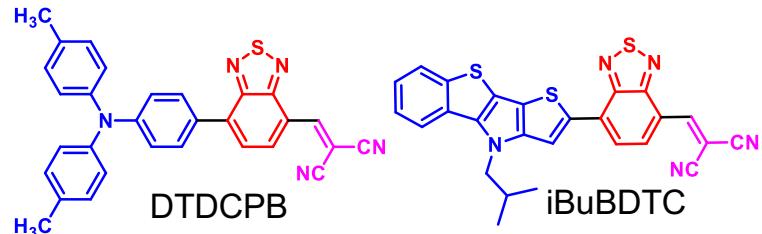
(c)



(d)



# Vacuum Deposited Dipolar Small Molecules in Biternary OPVs

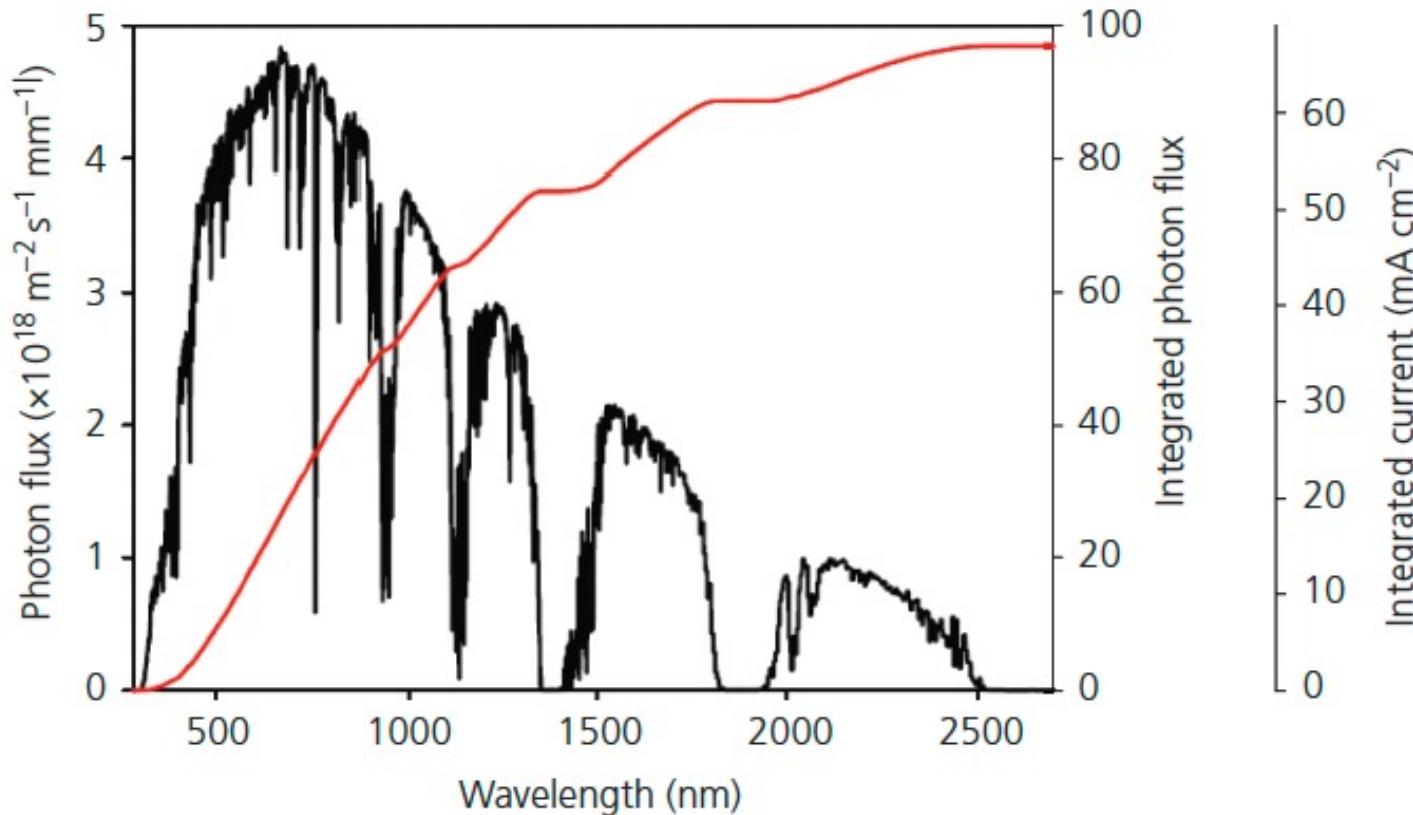


- Large dipole moment of d-a-a' donors evens the HOMO energies of the two when blended, allowing for continuous pathways for holes.
- Electrons conducted by common C<sub>70</sub> acceptor

Device	j <sub>sc</sub> [mA/cm <sup>2</sup> ]	V <sub>oc</sub> [V]	FF [%]	η <sub>P</sub> [%]
Binary (BTDC-Fo, 80 nm)	14.8	0.98	57.7	8.5
Binary (DTDCPB, 80 nm)	15.0	0.89	68.5	9.2
Bi-ternary (60 nm)	13.7	0.98	70.0	9.4

Li, et al. J. Am. Chem. Soc. 121, 18204 (2018)

# Designing Materials for OPVs

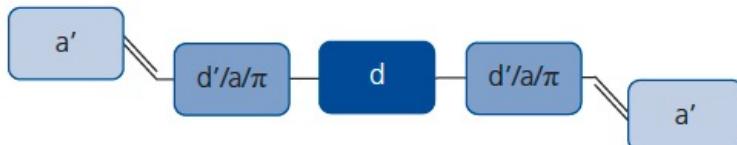


- Most of the usable (by single junction cell) spectrum is at  $\lambda < 1.5 \mu\text{m}$
- Thermodynamic limits are reached at  $\lambda \sim 1 \mu\text{m}$
- These considerations guide materials choices

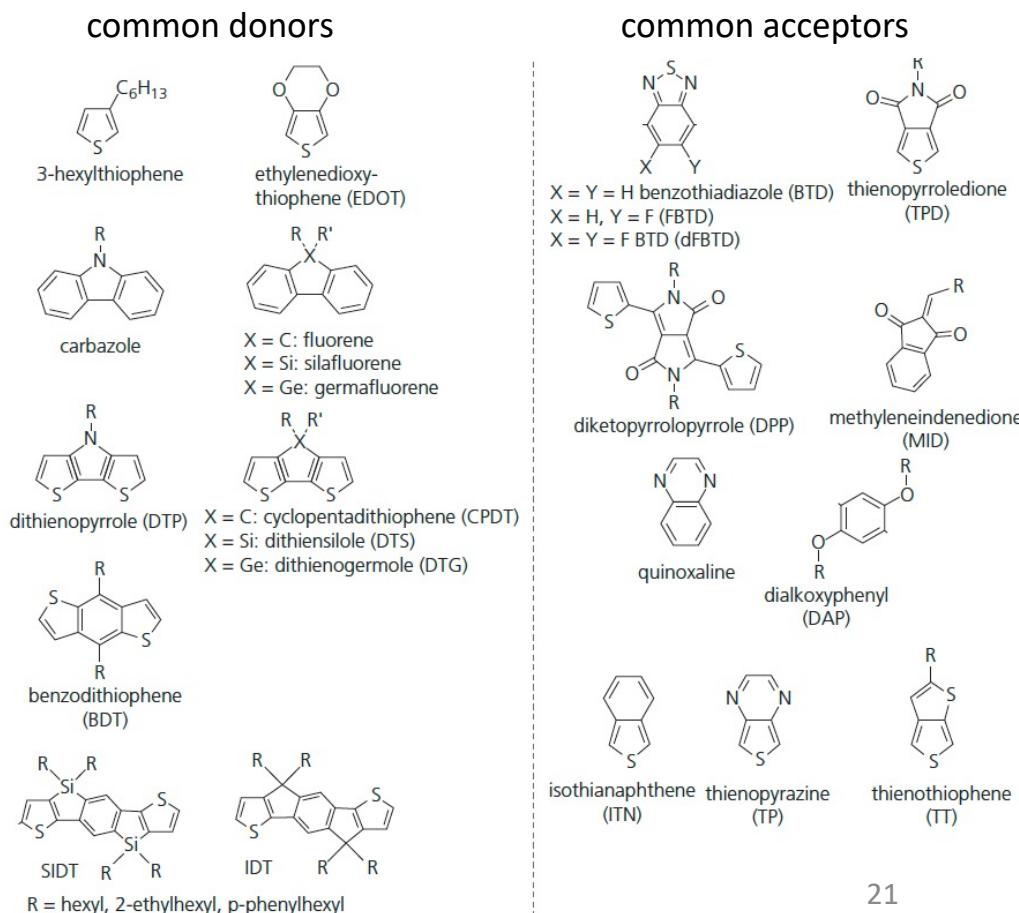
# Modular design of materials

- A method of functionalizing molecular cores to :
  - tailor wavelength coverage
  - improve morphology to achieve crystalline of appropriate scale
  - increase charge mobility

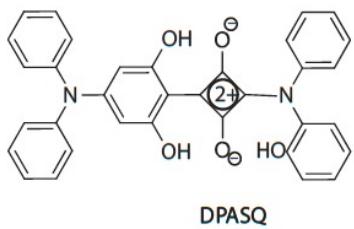
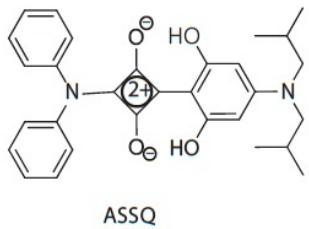
- Example: donor architecture



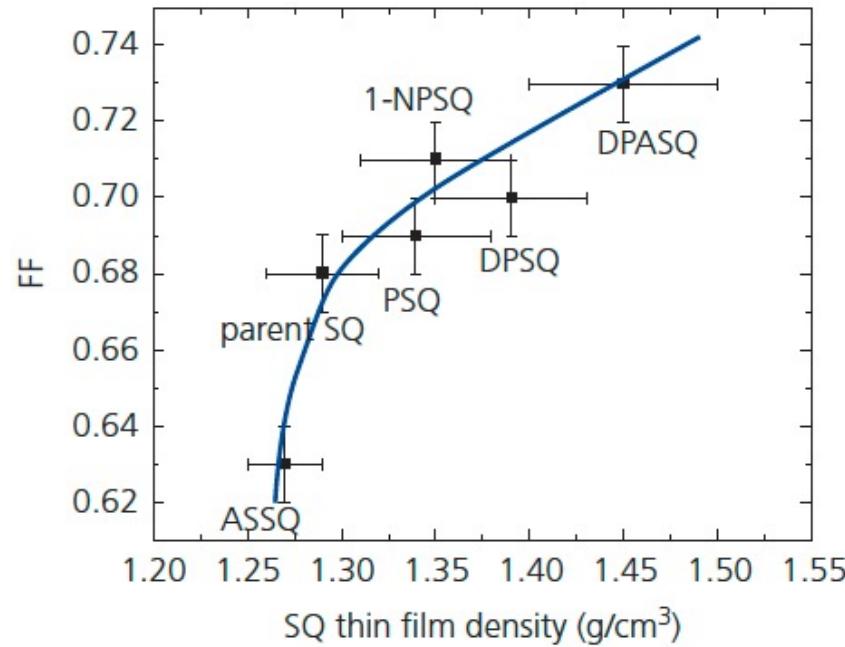
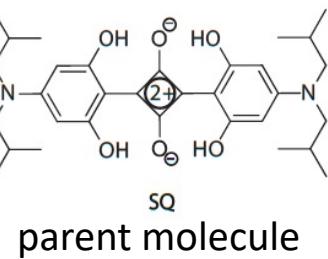
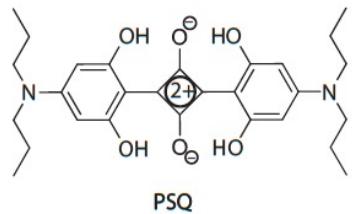
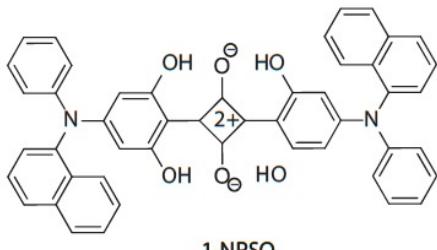
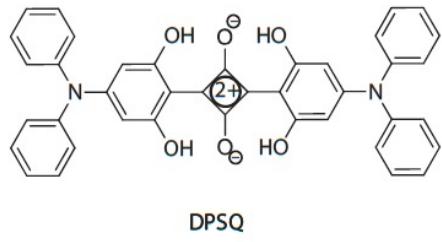
a=electron withdrawing (acceptor-like) moiety  
d=electron donating (donor-like) moiety  
π=bridge group



# Squaraine Donors: Functionalized to improve stacking

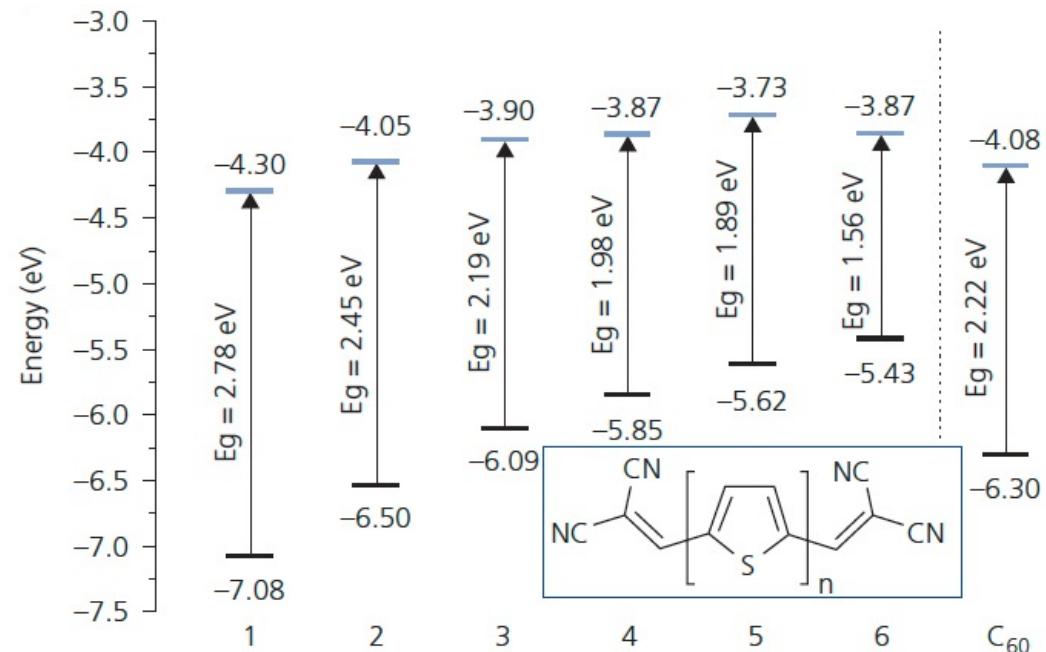
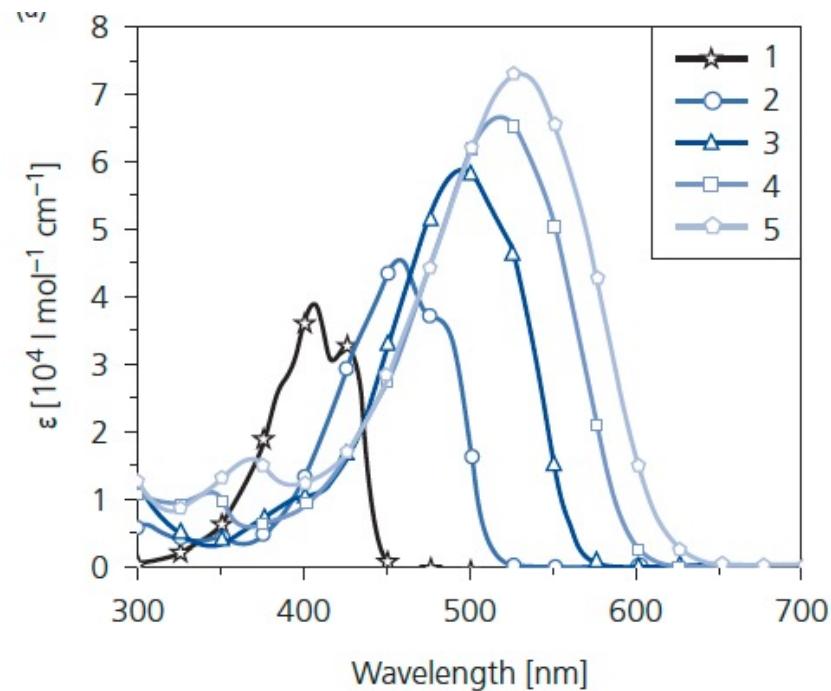


Phenyl groups promote planar  $\pi$ -stacking  
⇒ increased hole mobility  
⇒ increased fill factor



# Oligothiophenes: Extending the core to increase the conjugation length

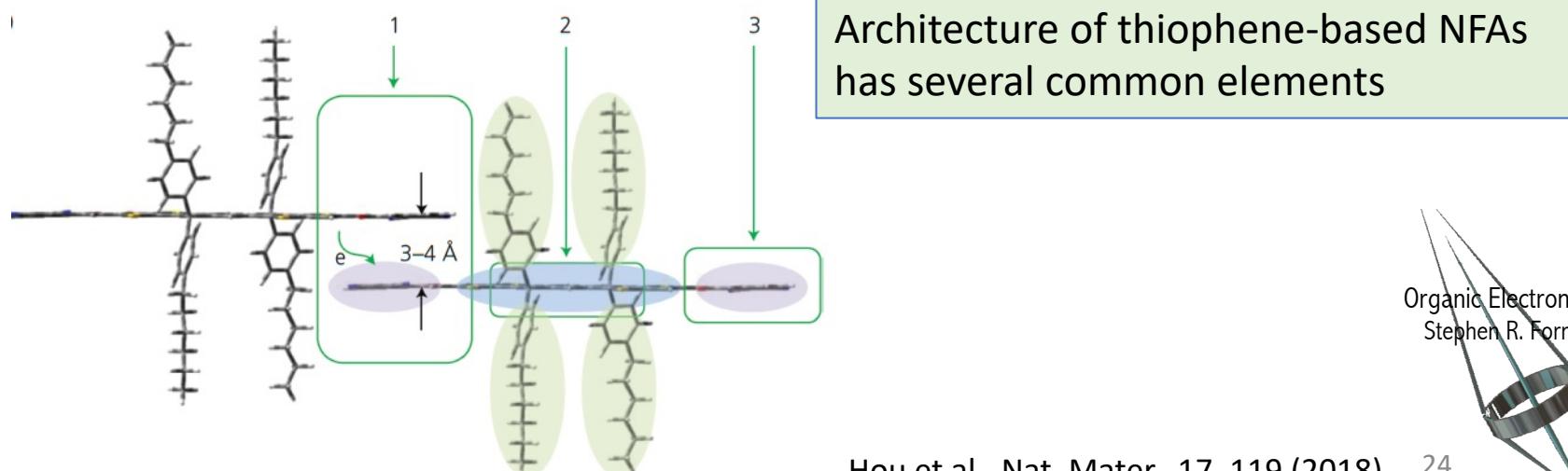
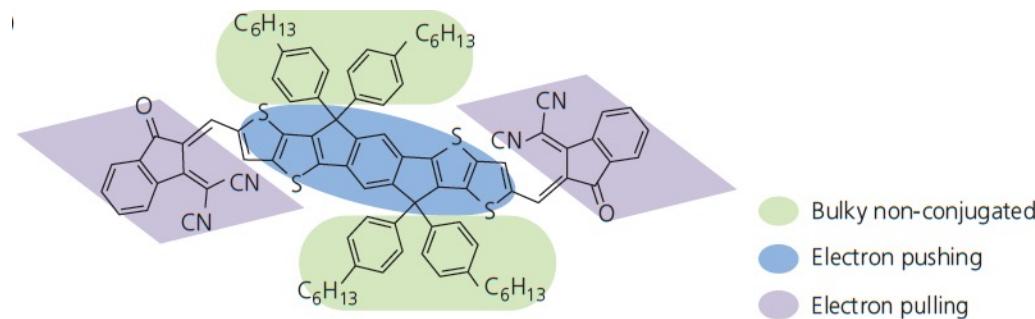
- Longer electron conjugation  $\Rightarrow$  lower excited state energy  
 $\Rightarrow$  red shift in absorption



# Non-Fullerene Acceptors

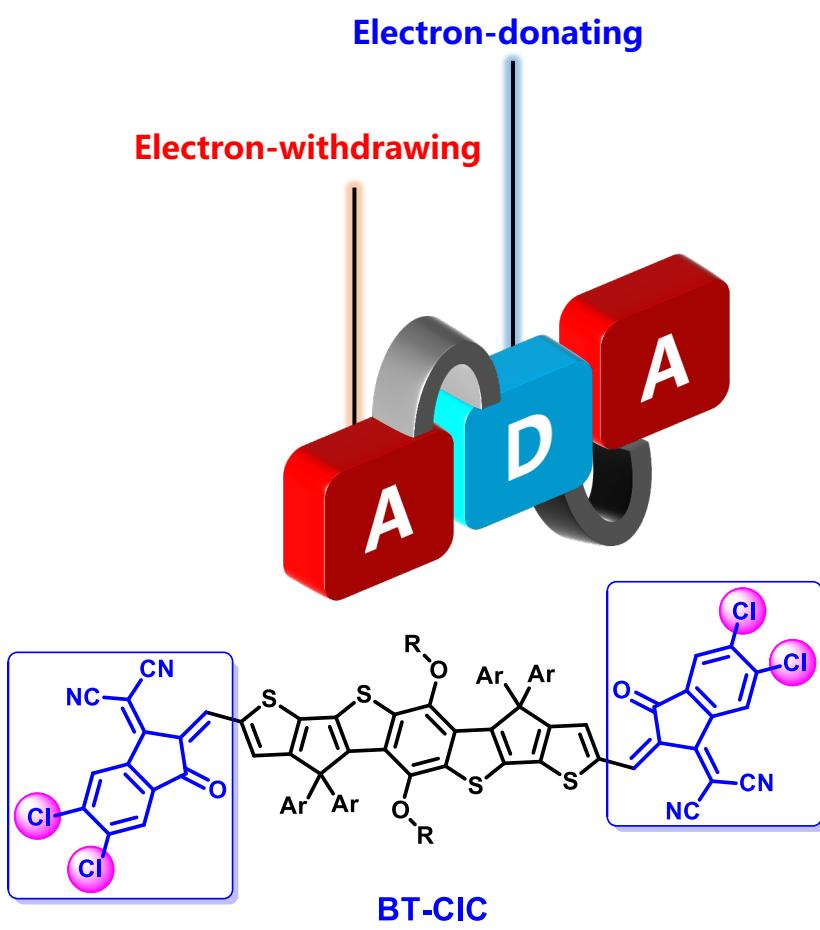
Molecules perfectly adapted for wavelength coverage and high conductivity

- Thiophene donor cores extended for increased conjugation
- Cores protected by alkane groups that also enhance solubility
- Acceptor end groups unprotected to enhance  $\pi$ -stacking

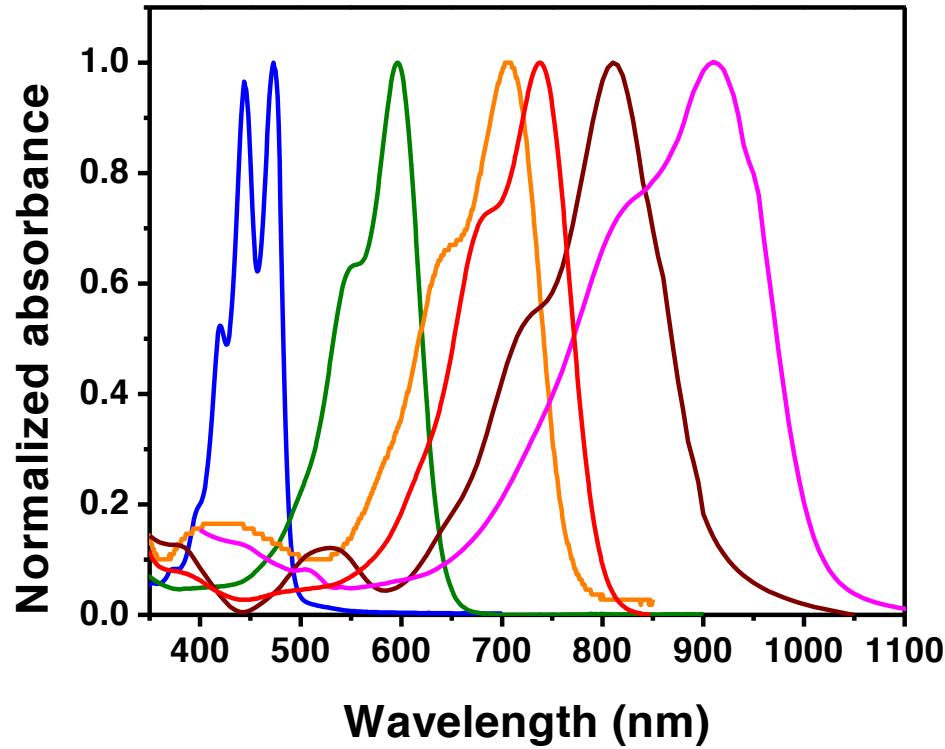


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# Non Fullerene Acceptor Design To Selectively Absorb Into the NIR

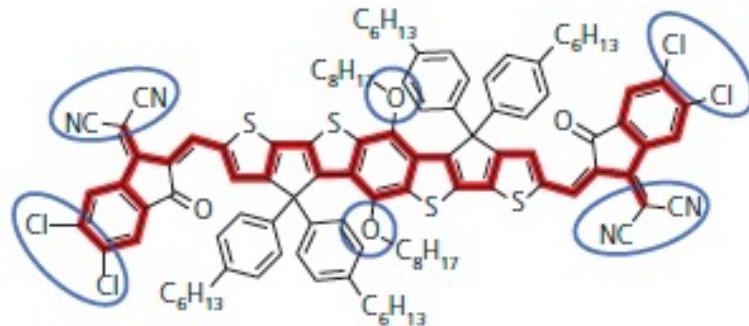


Absorption tunability presents opportunities unique to OPVs



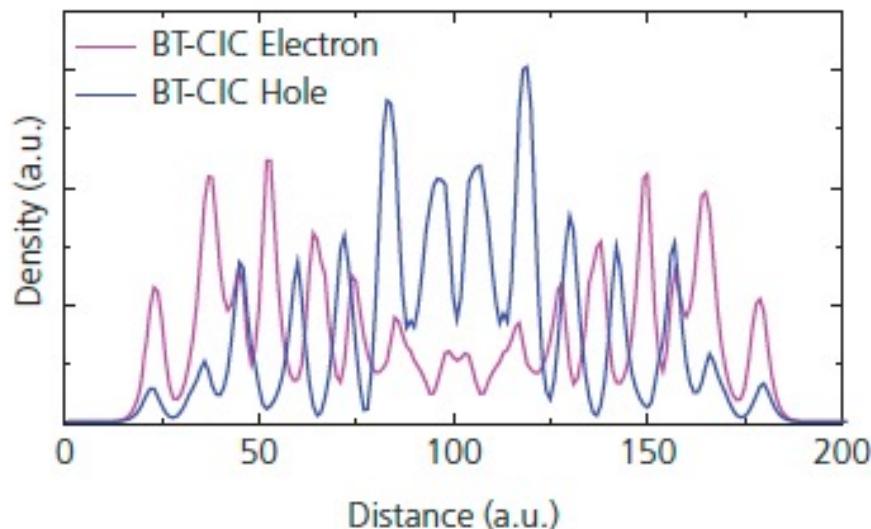
# Why does extending the core lead to red shifts?

(a)



Extending the core increases the electron conjugation length (red line)  
⇒ separation between electron and hole distributions increases  
⇒ reduced binding energy  
⇒ reduced energy loss

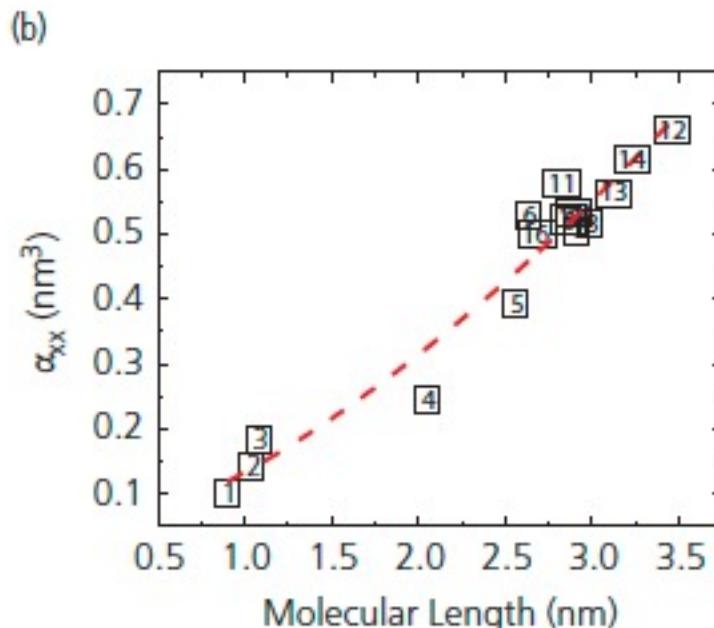
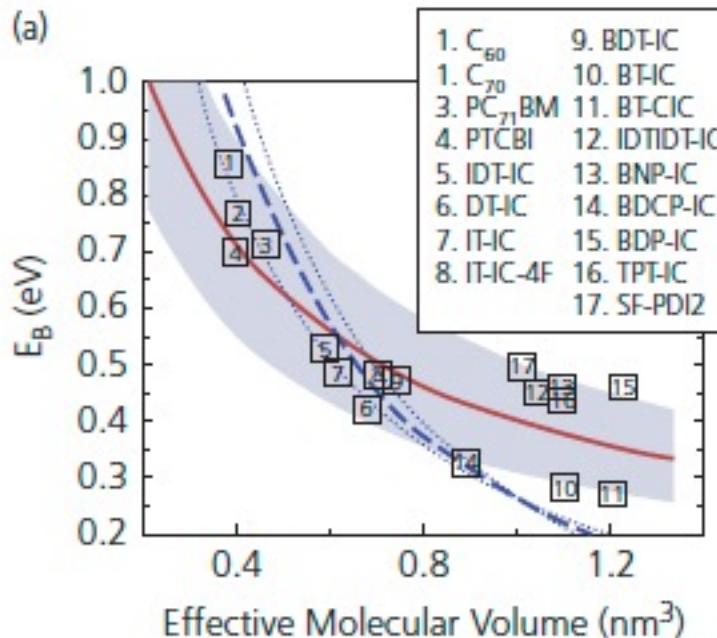
(b)



DFT calculation of electron and hole distributions along backbone

$$E_{loss} = E_X - qV_{OC}$$

# Binding energy of exciton a function of molecular “volume”



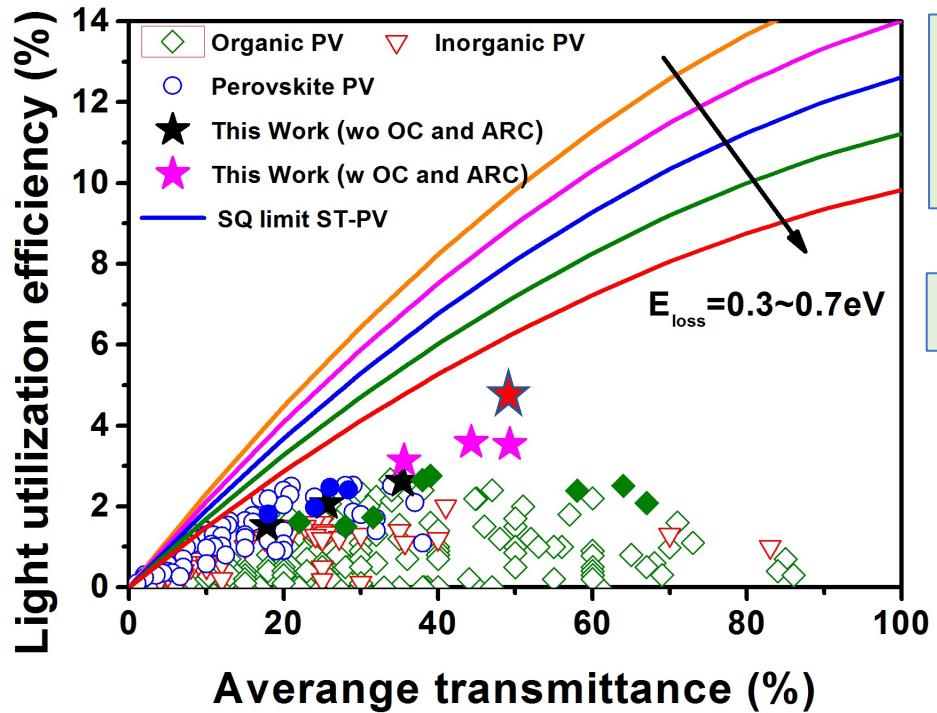
$$E_B = q^2 \int \frac{|\Psi_{0,0,0}(\mathbf{r}_h)|^2 |\Psi_{1,0,0}(\mathbf{r}_e)|^2}{4\pi\epsilon_0 f |\mathbf{r}_h - \mathbf{r}_e|} d^3\mathbf{r}_h d^3\mathbf{r}_e$$

A single value of dielectric constant ( $f$ ) fits all molecules (solid line)

$$\alpha_{xx} = \alpha_0 + \alpha_1 l^n$$

Molecular polarizability (related to  $f$ ) follows same trend with molecular length

# Wavelength-selective Absorption Can Lead to Semitransparent OPVs



Power generating windows (transparent in the visible, absorbing in the NIR) are a major opportunity unique to OPVs

Transparent OPV Figure of Merit

$$LUE = PCE \times APT$$

LUE: light utilization efficiency

PCE: power conversion efficiency

APT: average photopic transmission

$$APT = \frac{\int T(\lambda)P(\lambda)S(\lambda)d(\lambda)}{\int P(\lambda)S(\lambda) d(\lambda)}$$

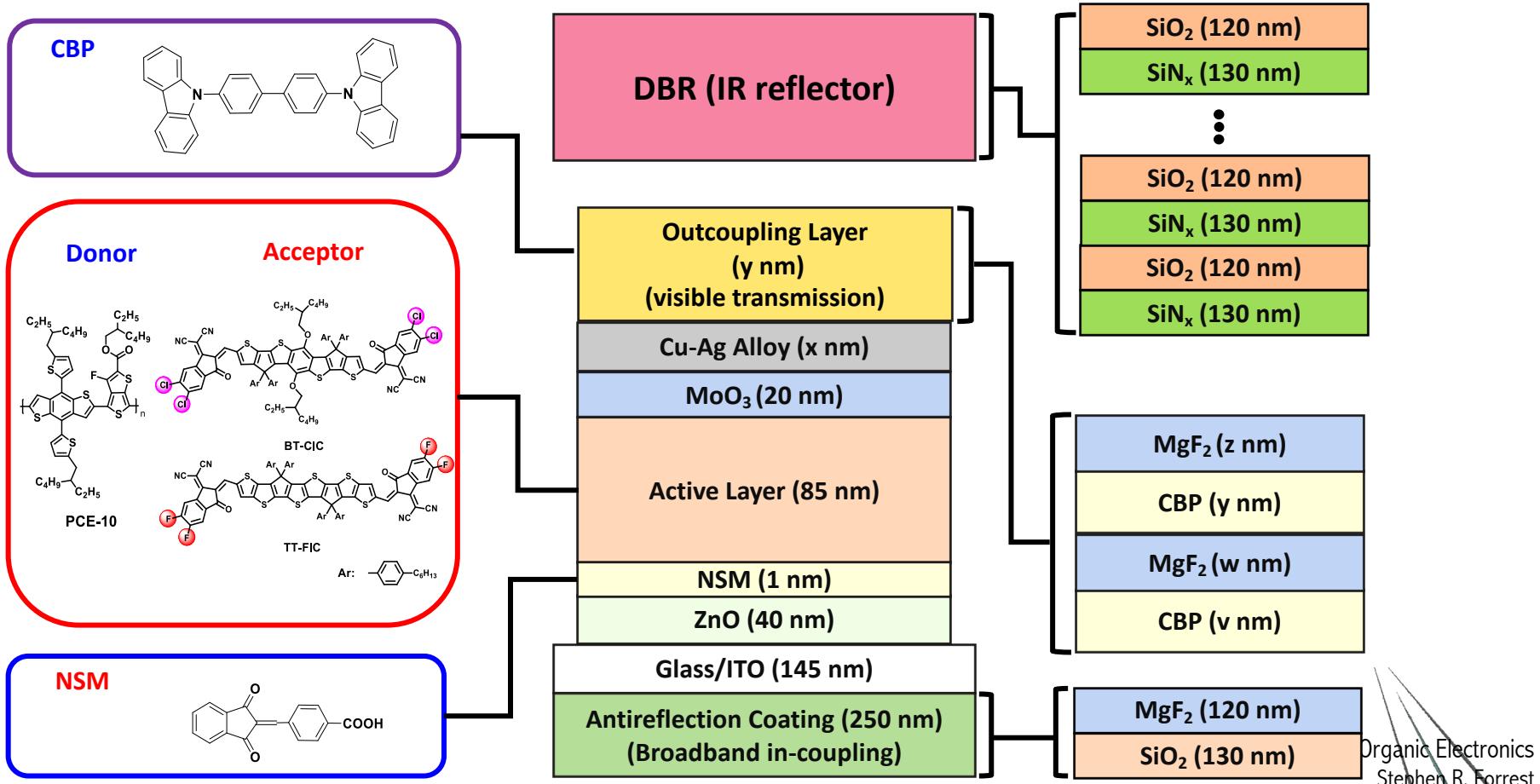
$\lambda$ : wavelength;

$T$ : transmission

$P$ : photopic response;  $S$ : solar photon flux (AM1.5G)

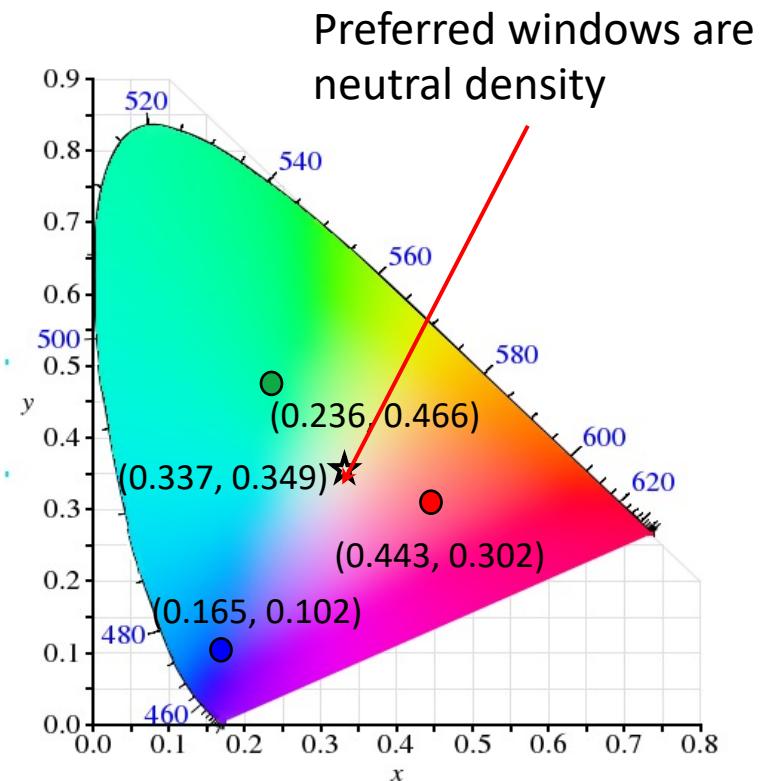
# Semi-Transparent Device Materials & Structures

## Example of Photon Management



# Semi-Transparent Organic Solar Cells

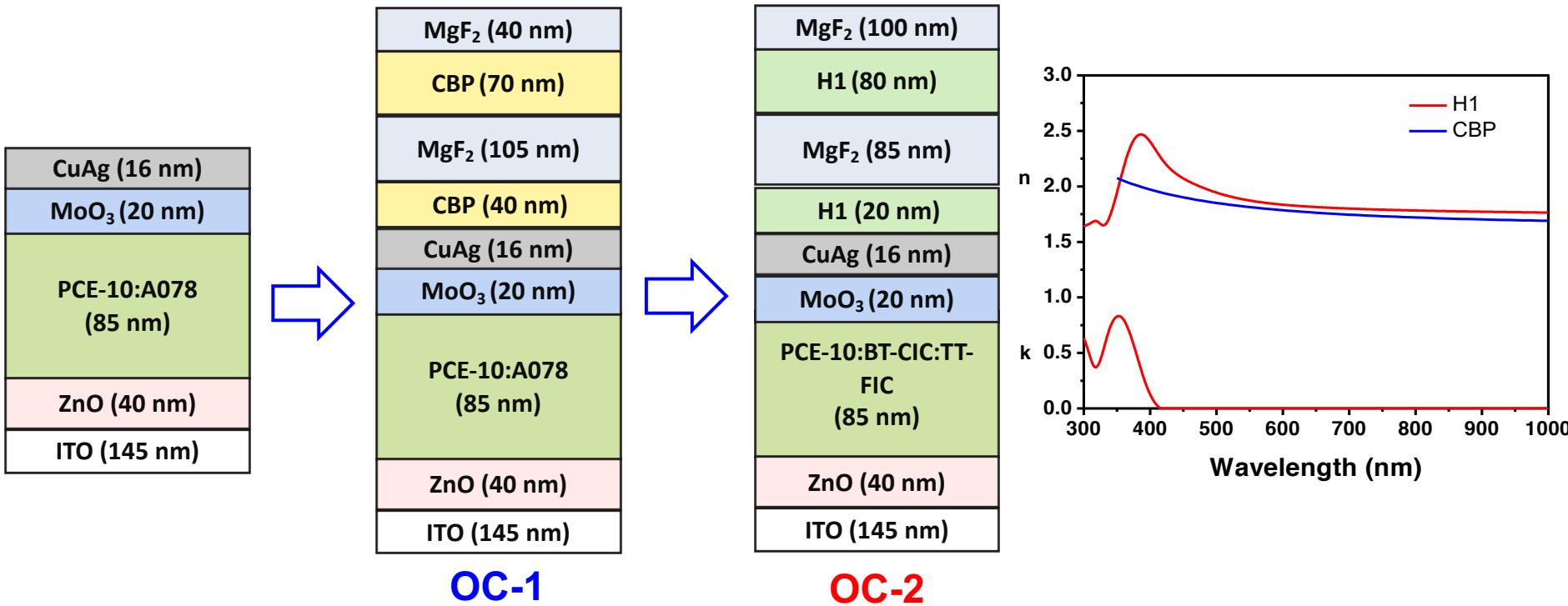
## Color Tunable Windows



Active Layer	$J_{sc}$ [mA/cm <sup>2</sup> ]	$V_{oc}$ [V]	FF [%]	PCE [%]
PCE-10:BT-CIC (1:1.5)	22.5 (21.3)	0.70	71.0	11.2
PCE-10:DTD-FIC (1:1.5)	25.6 (21.3)	0.64	69.5	11.3

Electronics  
in R. Forrest

# Optical Outcoupling Layers



Device	$J_{sc}$ [mA/cm <sup>2</sup> ]	$V_{oc}$ [V]	FF	PCE [%]	APT [%]	LUE [%]
WO OC	18.4	0.72	0.62	8.3	30.2	2.51
W OC-1	18.1	0.74	0.65	8.7	48.3	4.20
W-OC-2	-	-	-	-	51.2	5.2

