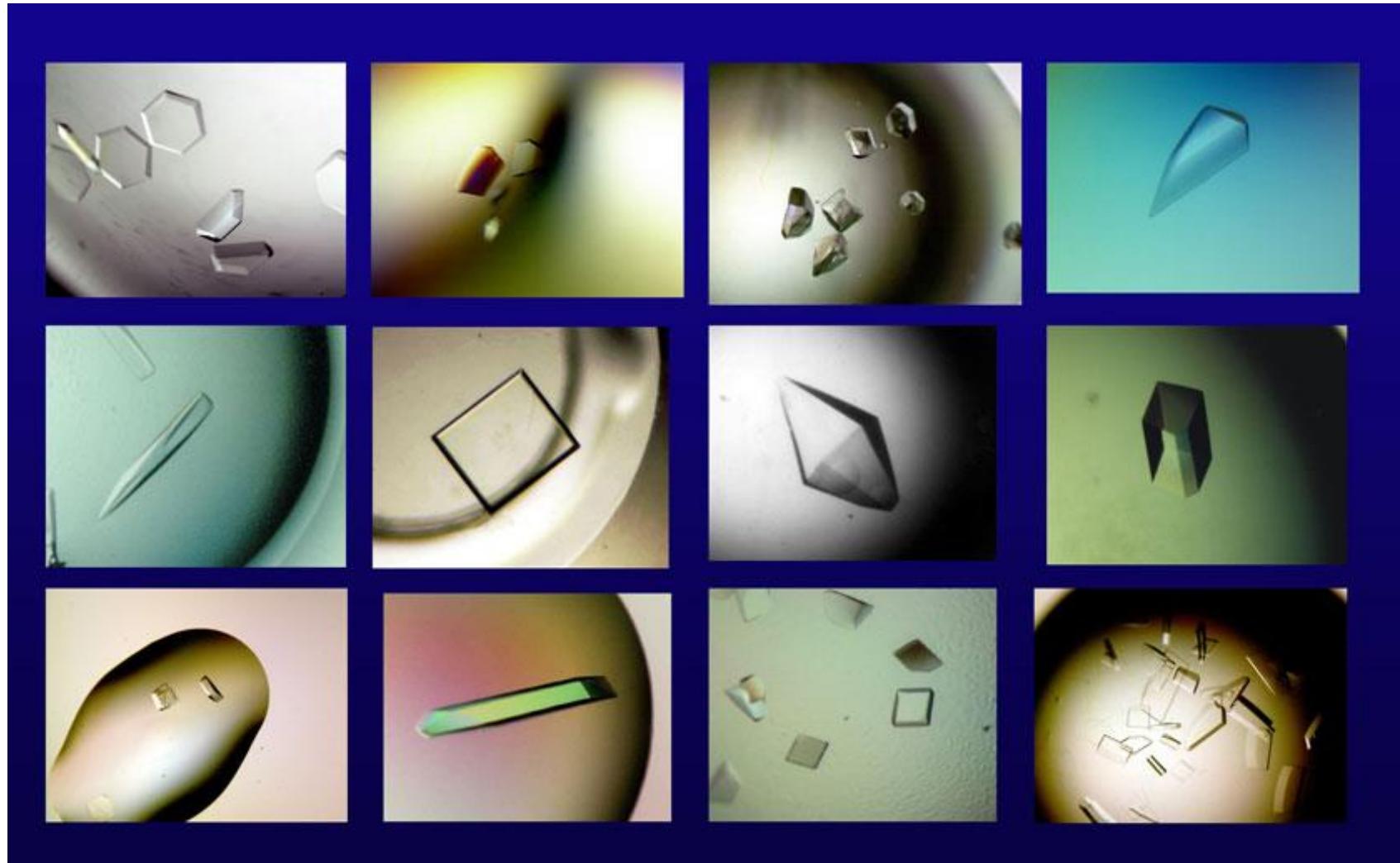
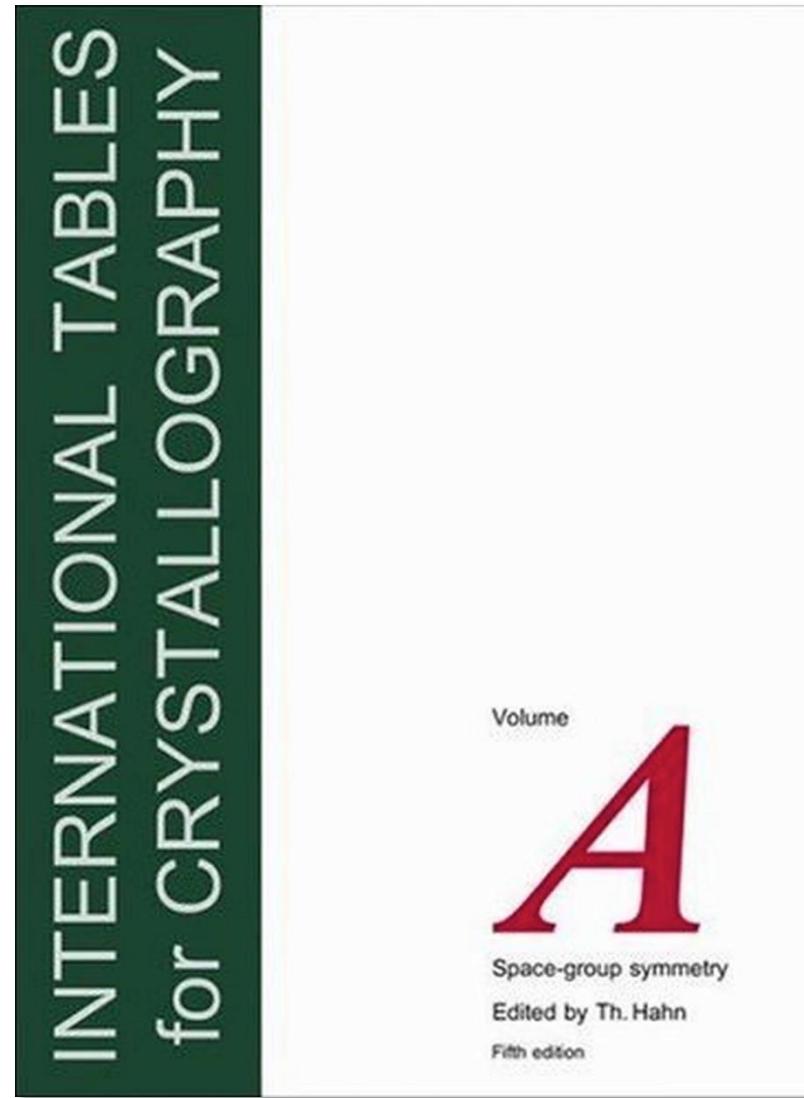


# Introduction to space groups

Andrey Lebedev, CCP4



Further referred  
to as ITC-A

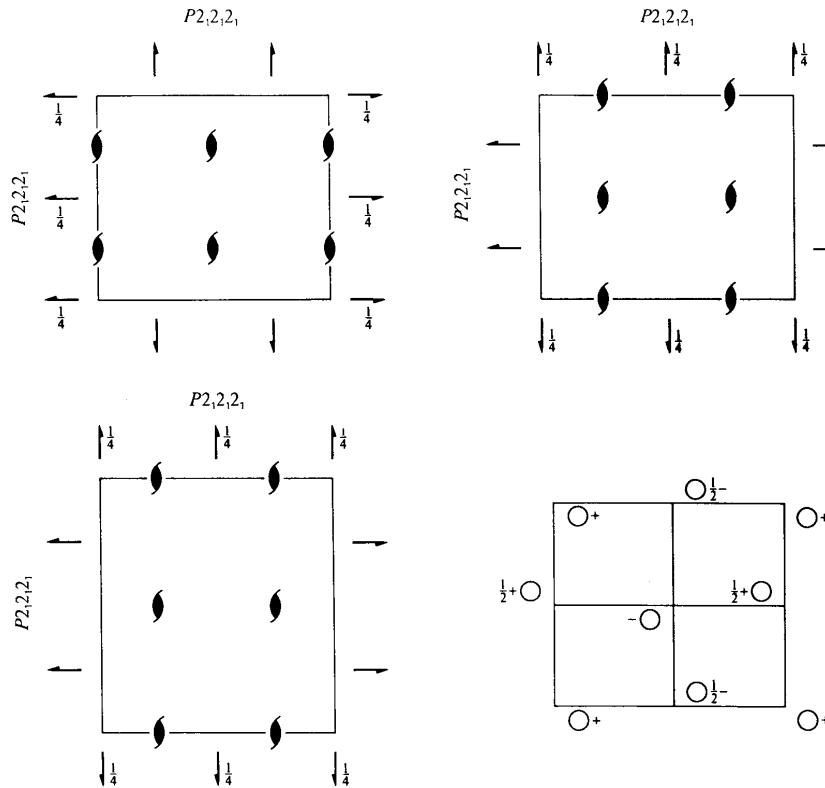


# Space group representation in ITC-A

$P2_1 2_1 2_1$   
No. 19

$D_2^4$   
 $P2_1 2_1 2_1$

222  
Orthorhombic  
Patterson symmetry  $Pmm$



INTERNATIONAL TABLES  
for CRYSTALLOGRAPHY  
Volume A  
Space-group symmetry  
Edited by Th. Hahn  
Fifth edition

# Space group representation in ITC-A

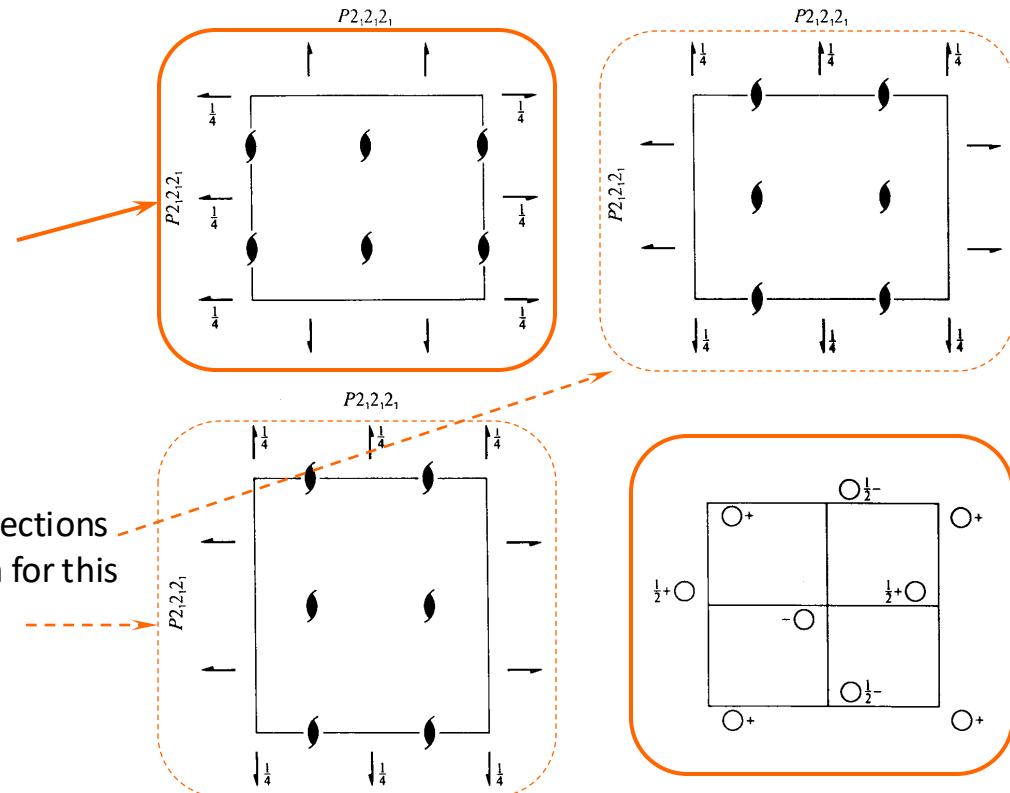
$P2_1 2_1 2_1$   
No. 19

$D_2^4$   
 $P2_1 2_1 2_1$

222

Orthorhombic  
Patterson symmetry  $Pmm$

Location of symmetry elements

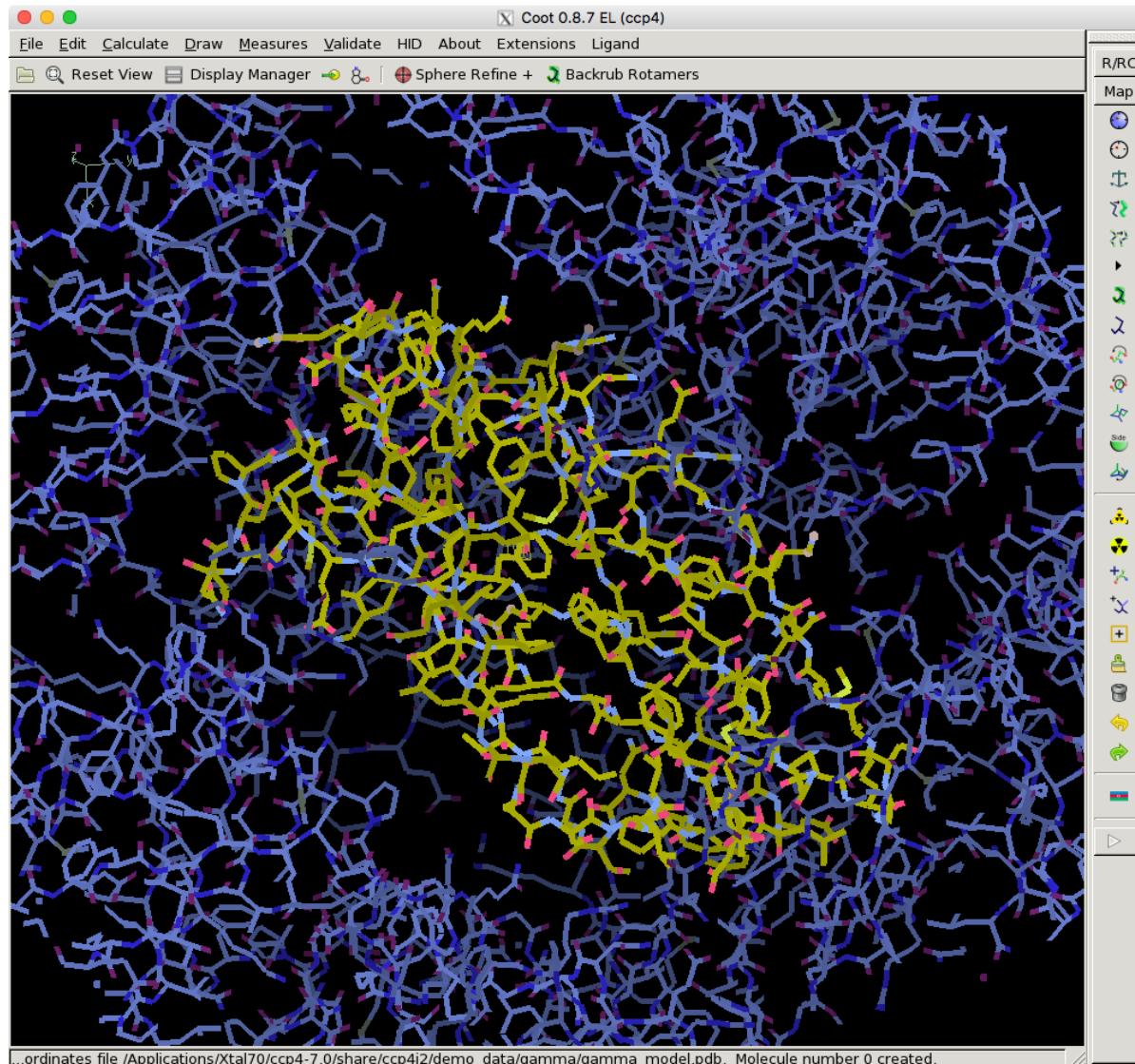


Two other projections are also shown for this space group

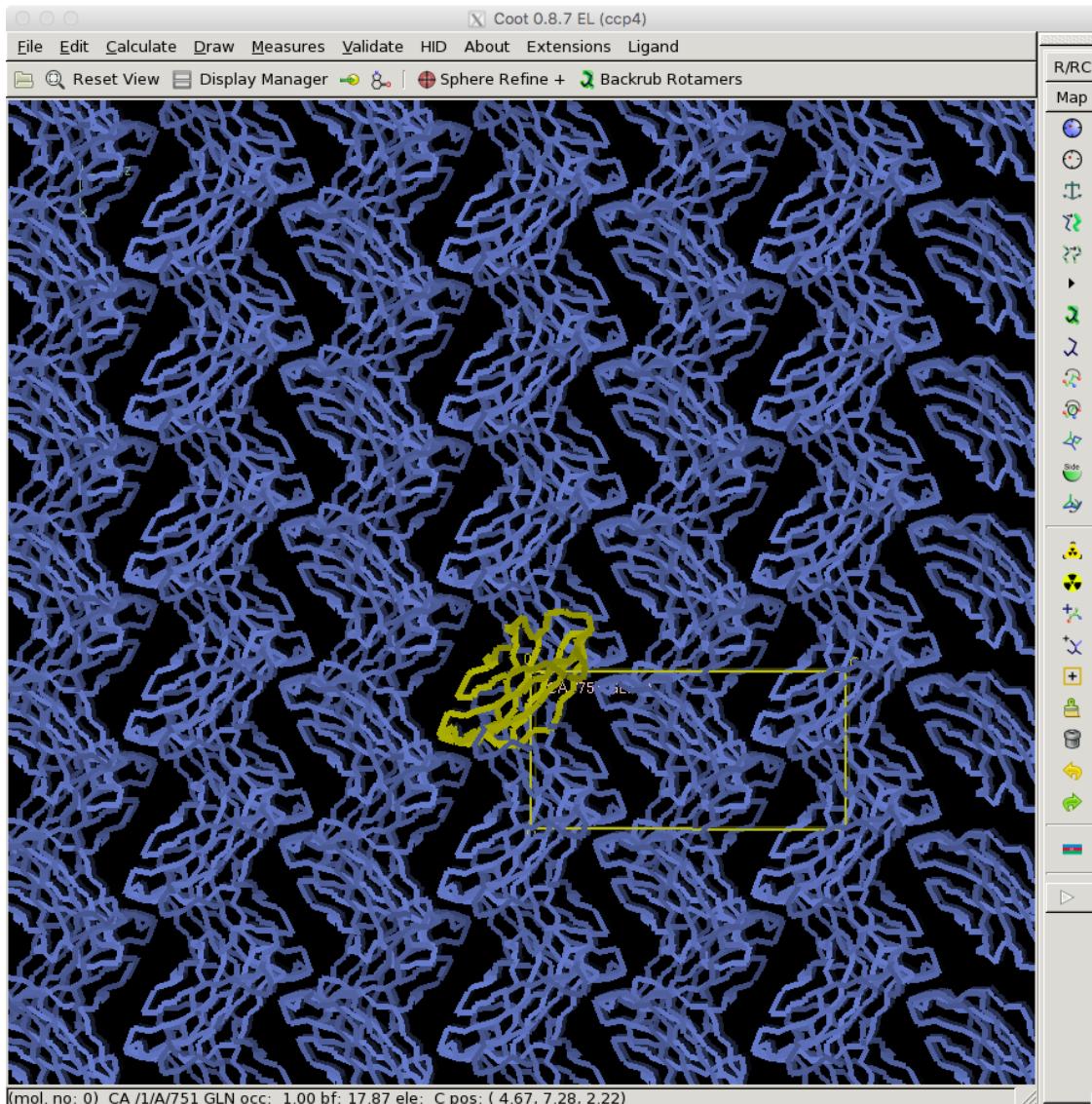
Set of equivalent points in general position.

We will be looking at "molecular wallpaper" instead

# Examine structure in Coot

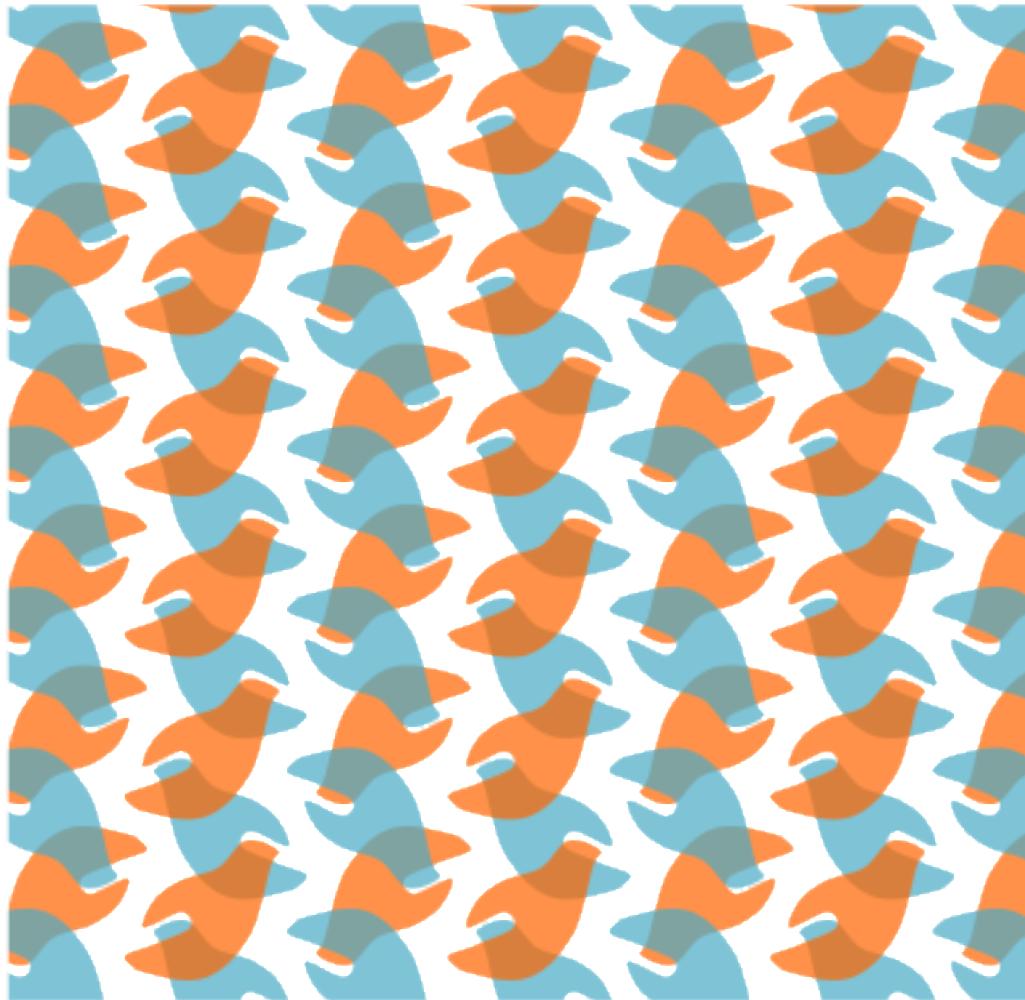


# Symmetry view in Coot



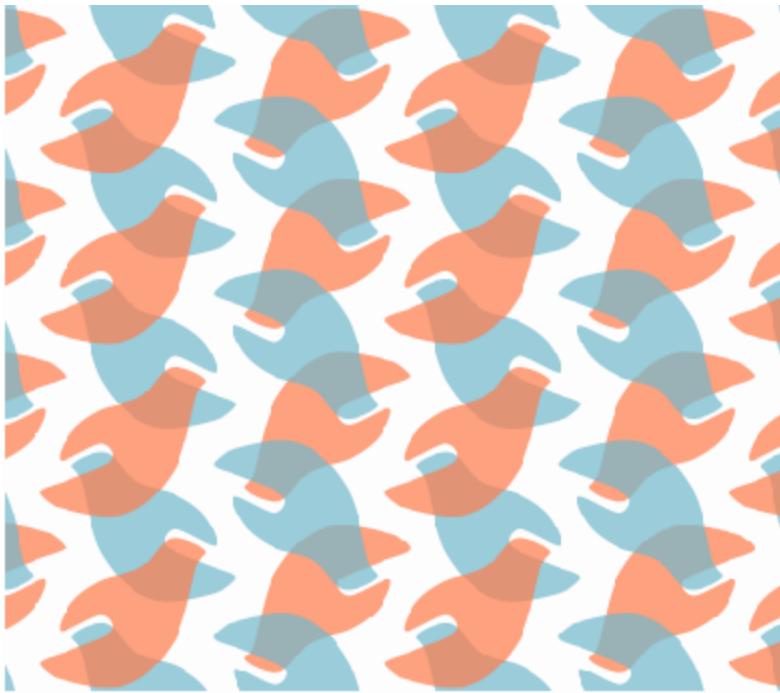
- Helps learning – compare your structure with the space group scheme in the International tables
- Helps building correct perception of what is the physical object and what is an “annotation”.  
(e.g. molecules are real, and unit cell is just a label; crystal is built of molecules not of unit cells!)

# Simplified representation



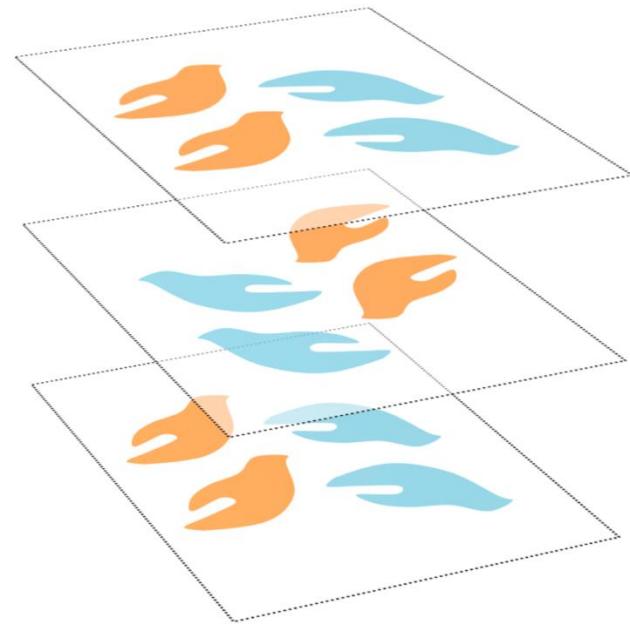
Each “molecule” has two sides,  
blue and orange

# Simplified representation

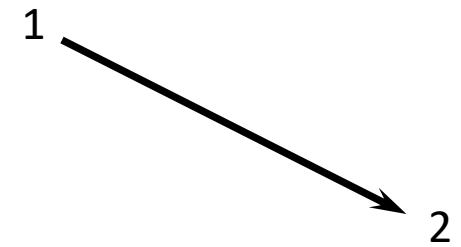
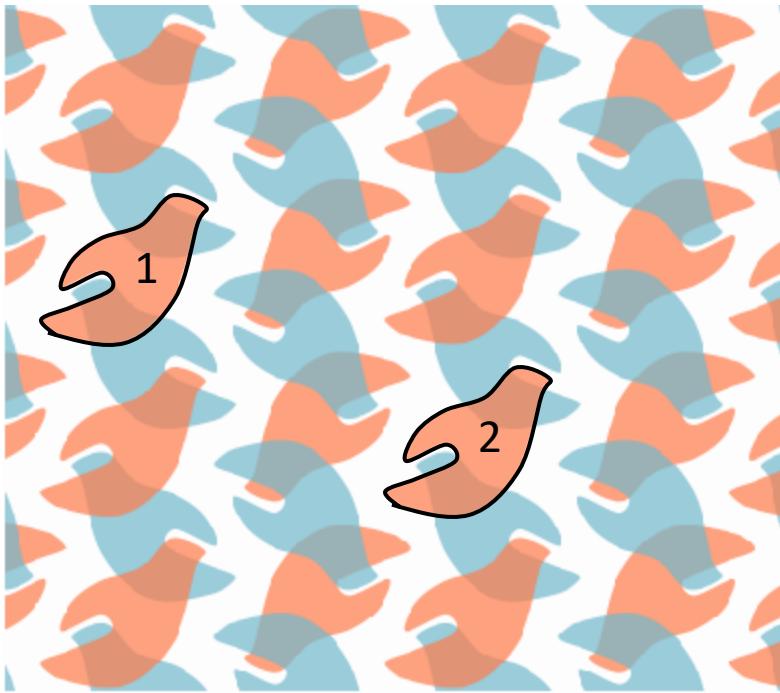


View from the top

There is a third dimension.

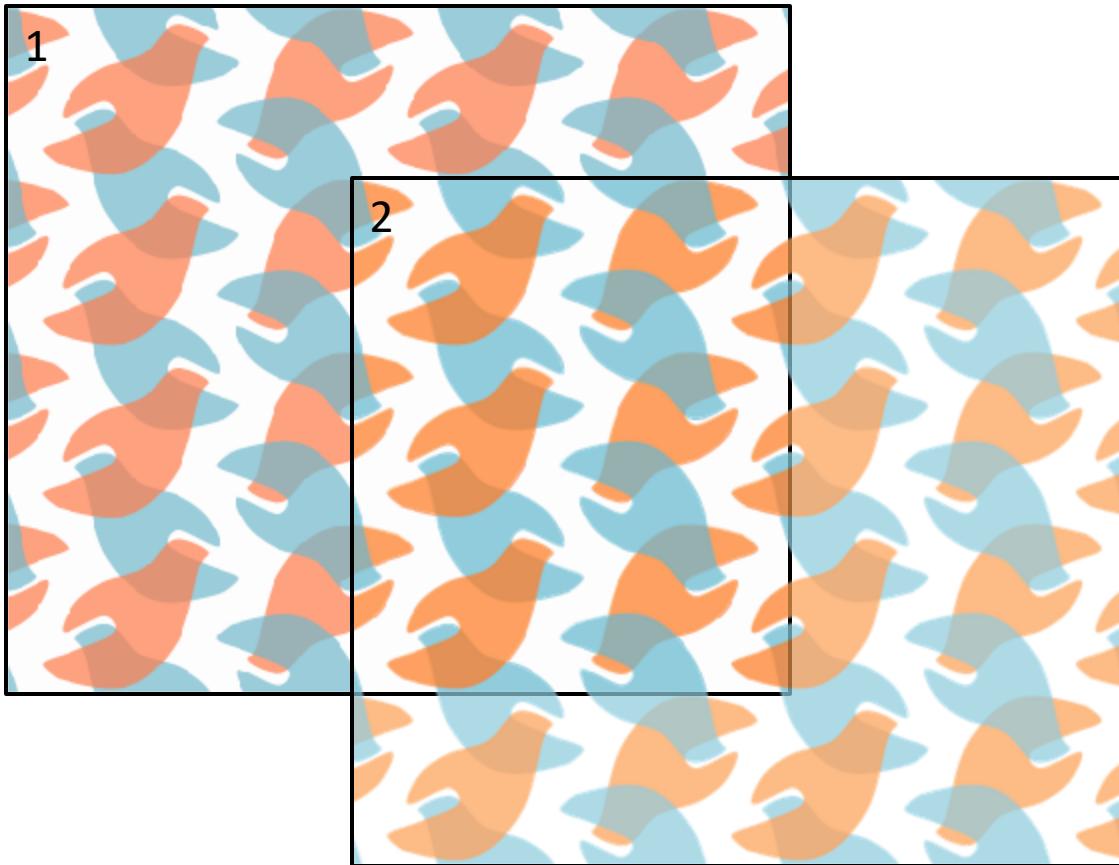


# Translation 1-2



Vector maps 1 → 2

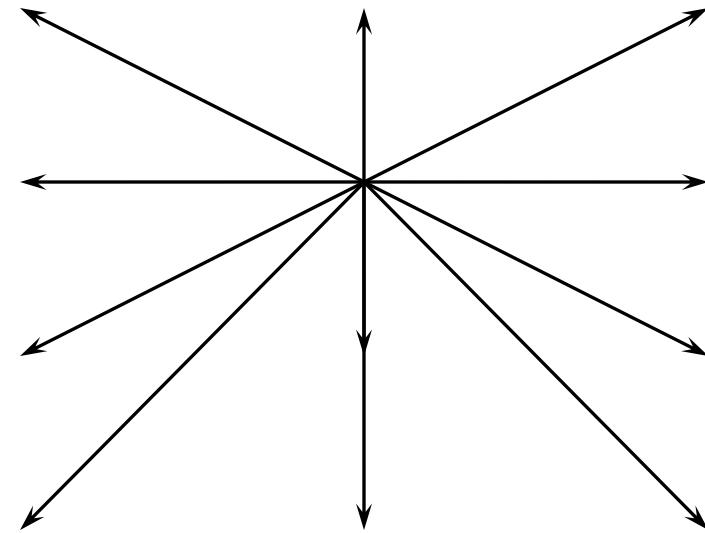
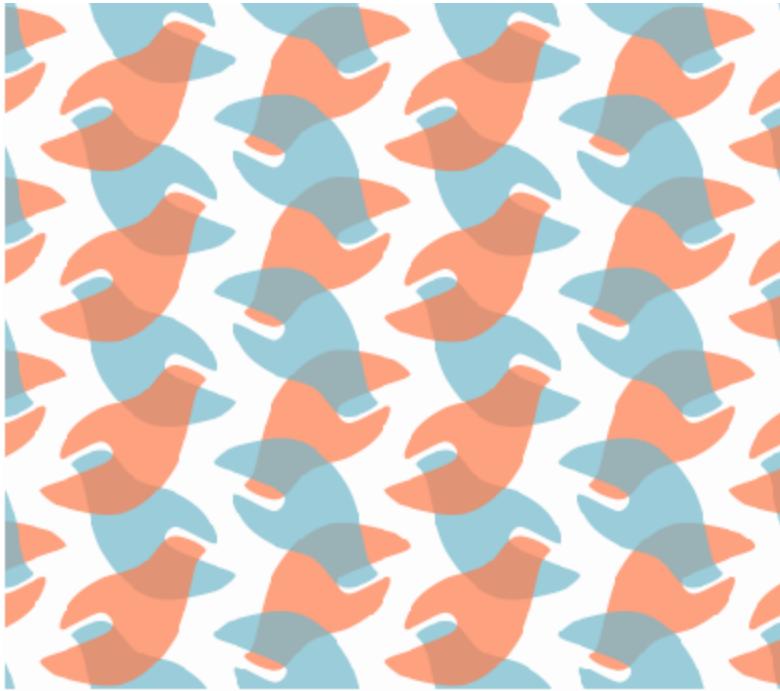
# Translation 1-2 is global



Vector  $1 \rightarrow 2$  maps  
the whole crystal  
onto itself:

it defines a  
**crystallographic**  
operation

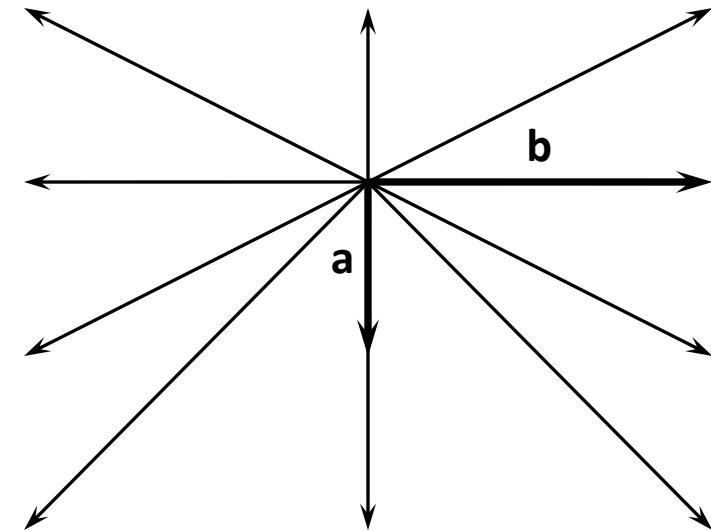
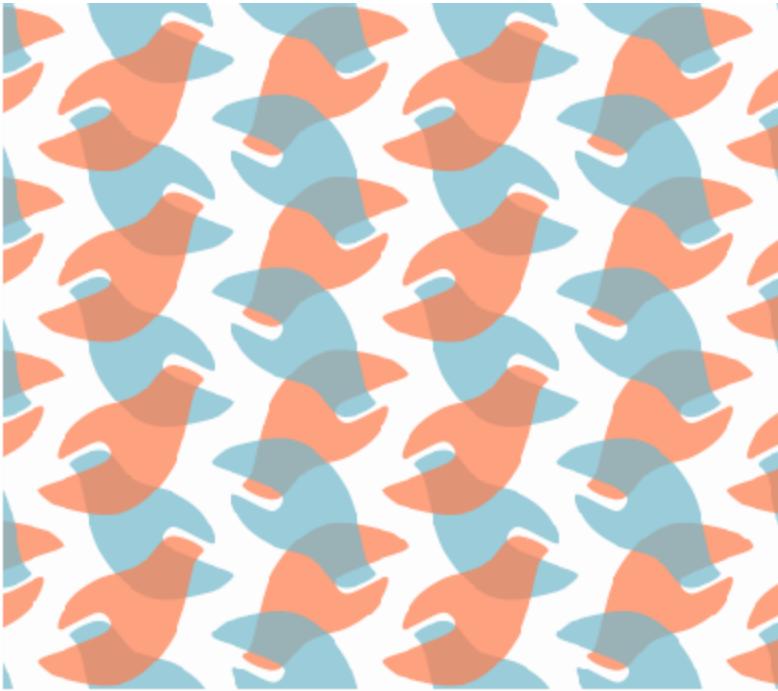
# All translations form an infinite group



An infinite group (over vector sum):

- inverse translations included
- sum of any two vectors from the group belongs to the group

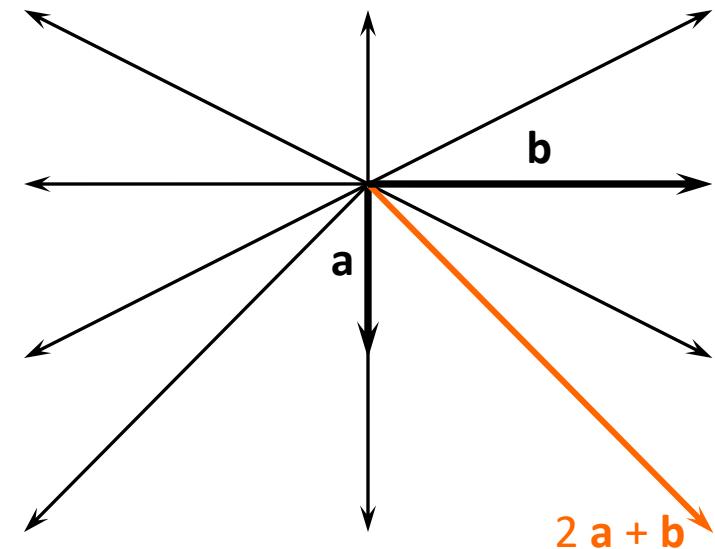
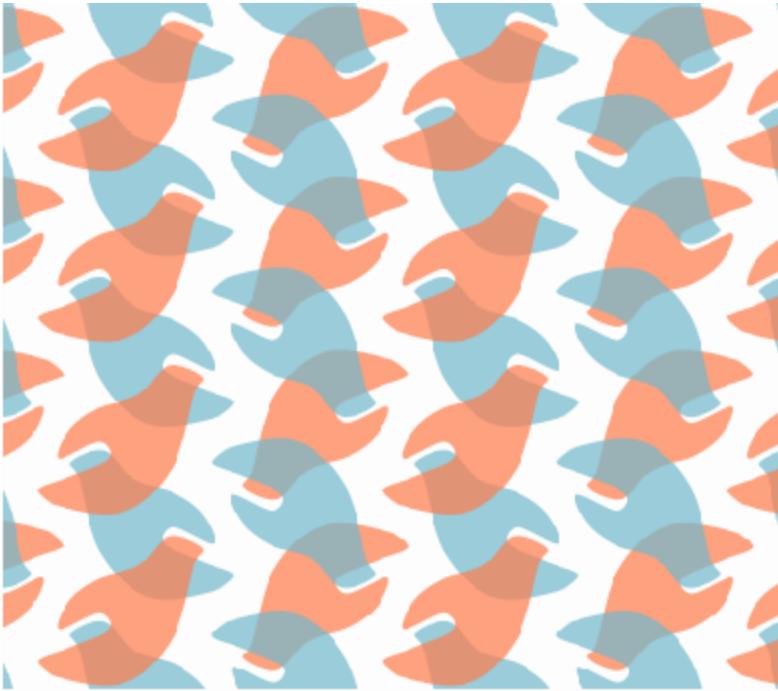
# Basis vectors



All the translations that map the crystal onto itself can be produced from a basis set: **a**, **b**, **c**

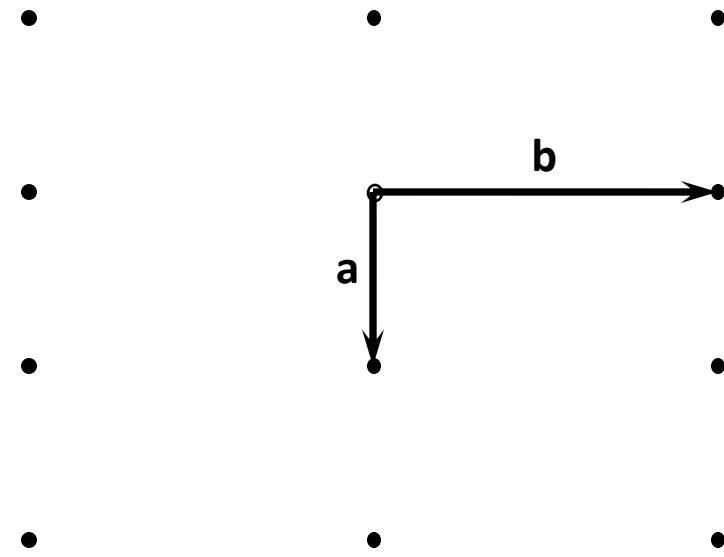
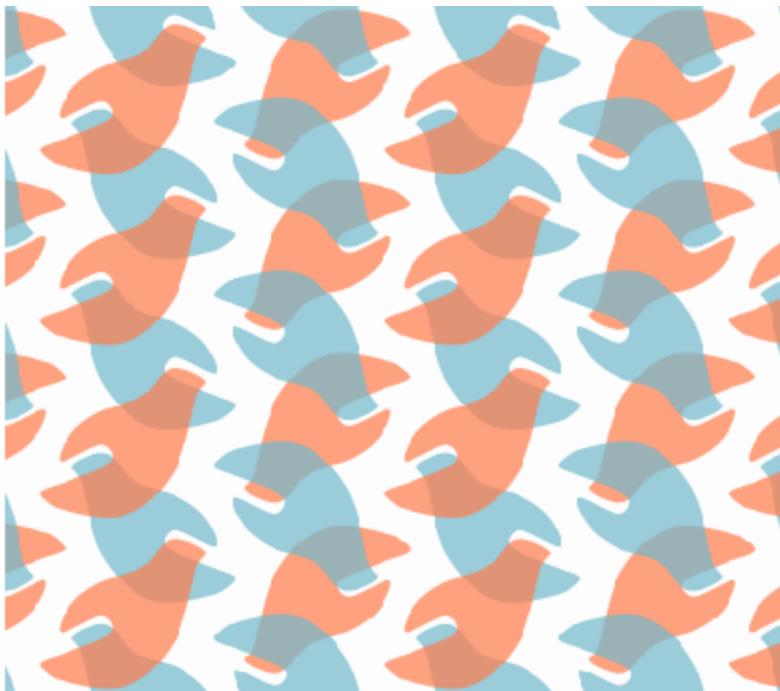
(**c** is perpendicular to the plane)

# Basis vectors



For example, the highlighted vector is expressed as  $2\mathbf{a} + \mathbf{b}$ .

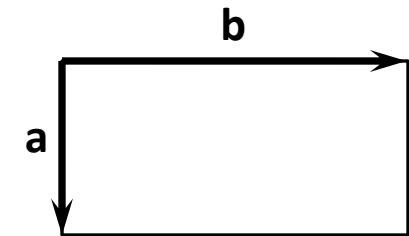
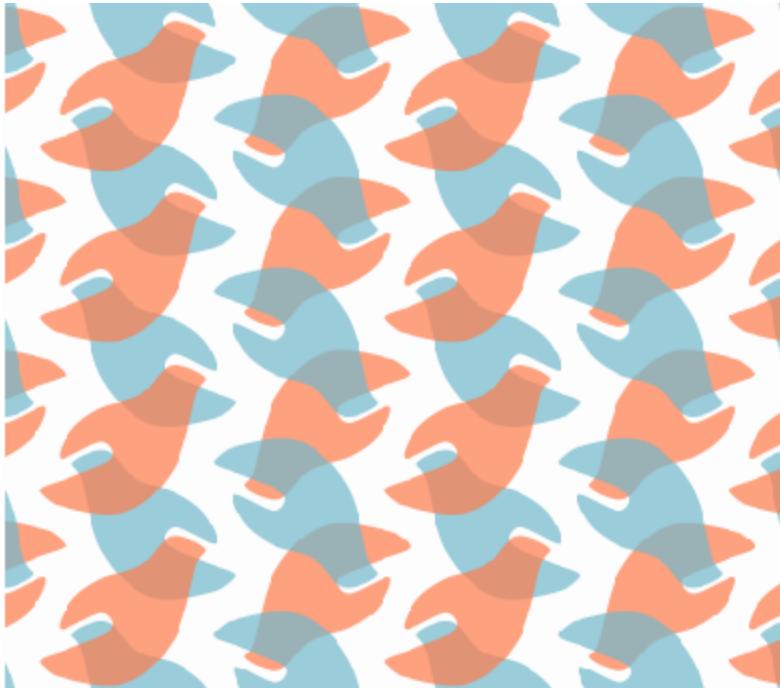
# Lattice



All the crystallographic translations can be represented as a lattice.

Translations, crystal lattice and unit cell live in a separate space.  
Coordinates in that space do not define coordinates of atoms but relation between equivalent atoms

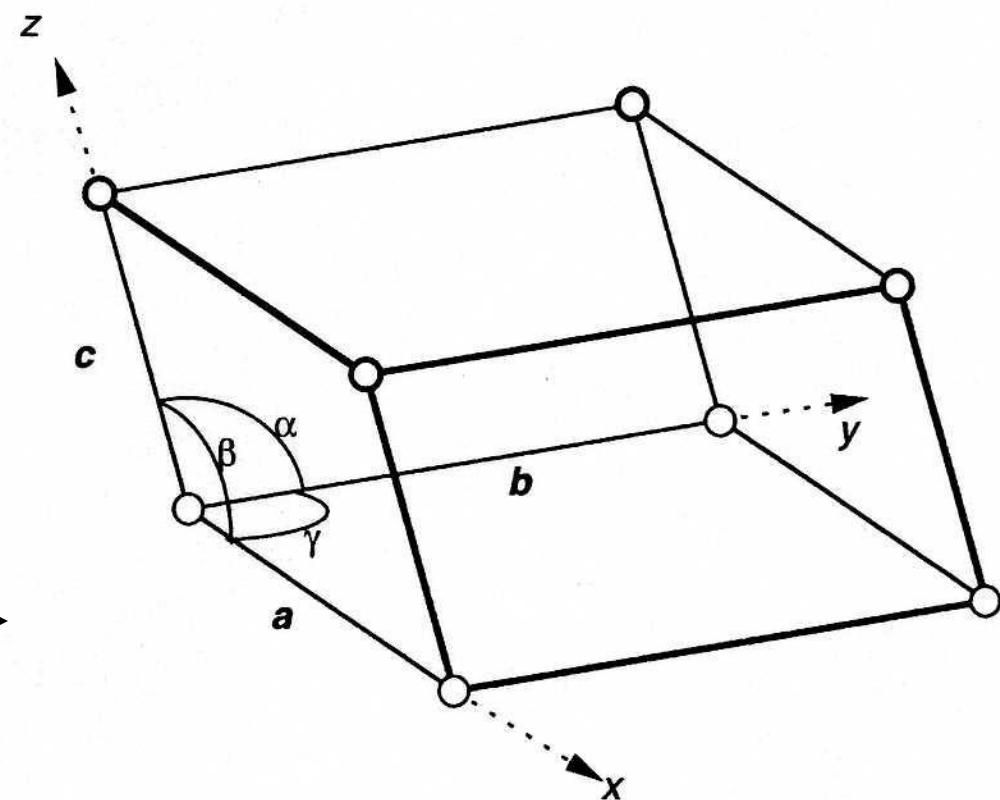
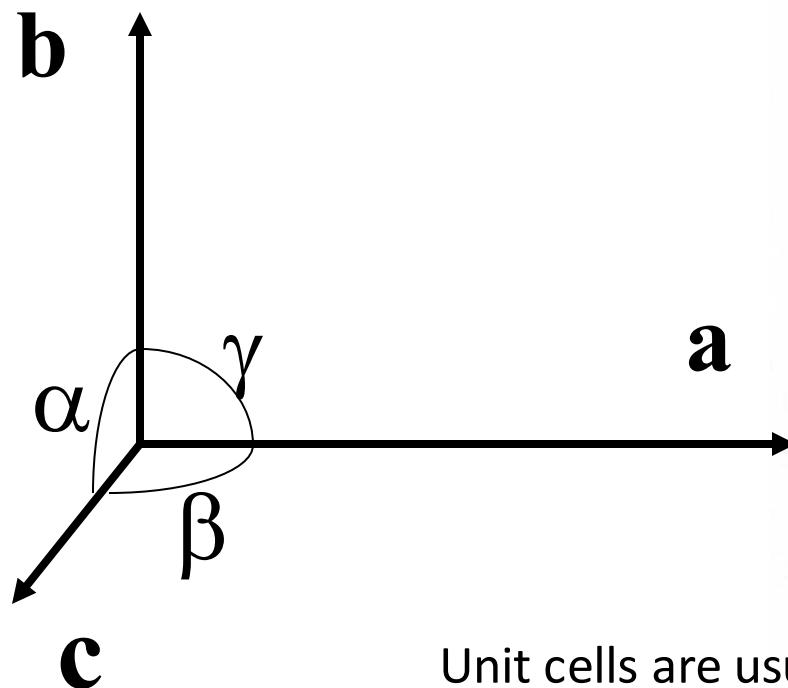
# Unit cell



A compact representation of translational symmetry and base vectors.

# Unit cell parameters (3D view)

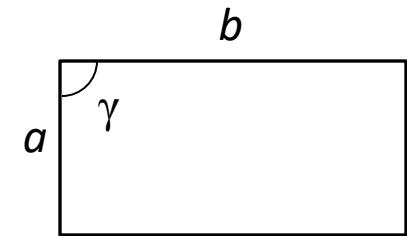
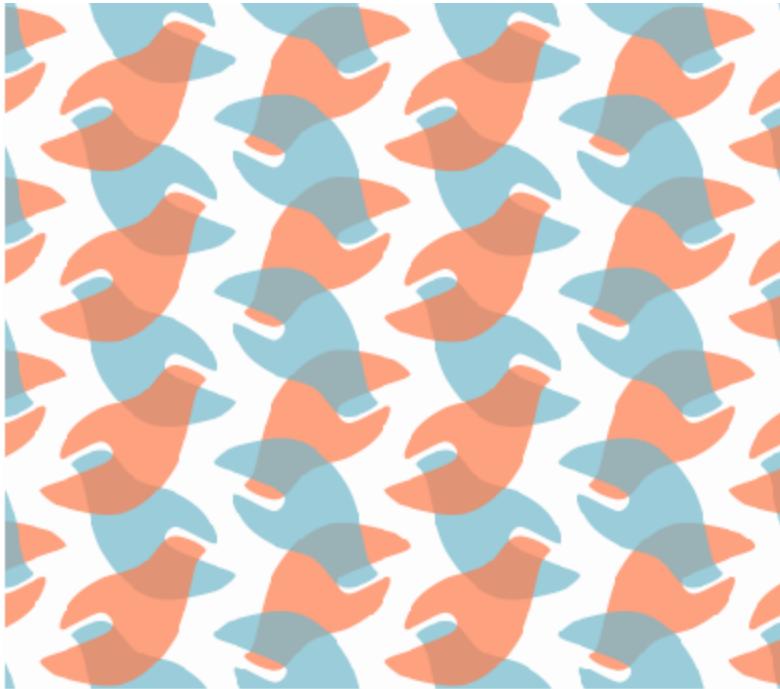
Translation symmetry is defined by three base vectors **a**, **b**, and **c**.



Unit cells are usually defined in terms of the *lengths* of these vectors and angles between them. For example,

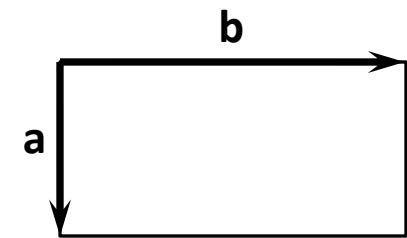
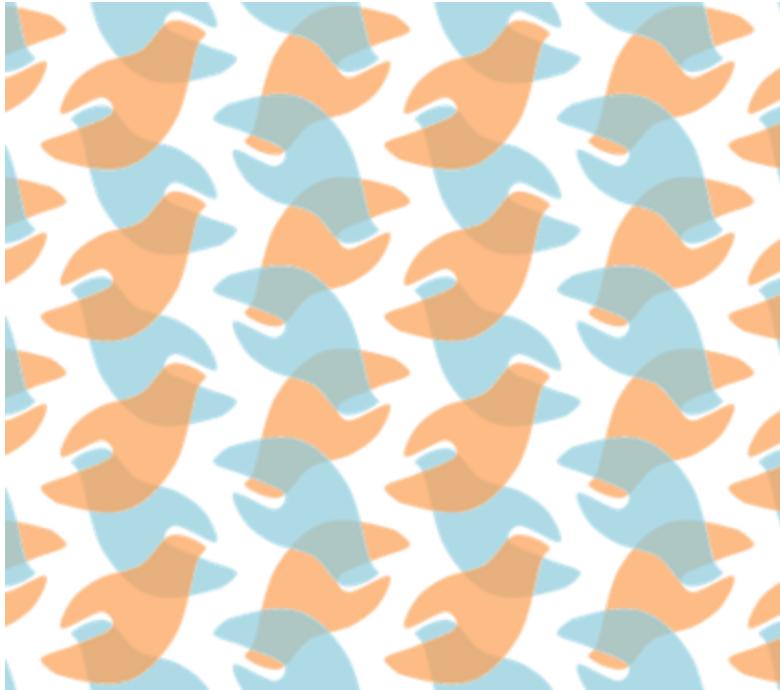
$$a=94.2\text{\AA}, b=72.6\text{\AA}, c=30.1\text{\AA}, \alpha=90^\circ, \beta=102.1^\circ, \gamma=90^\circ.$$

# Unit cell

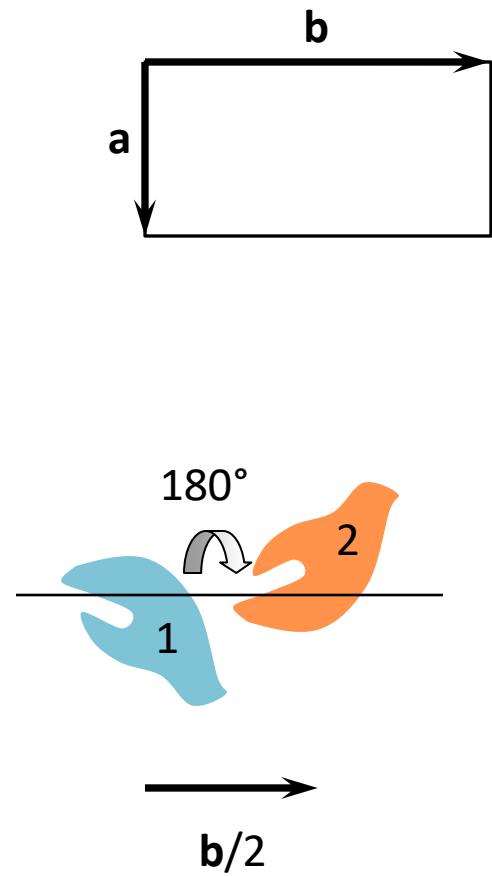
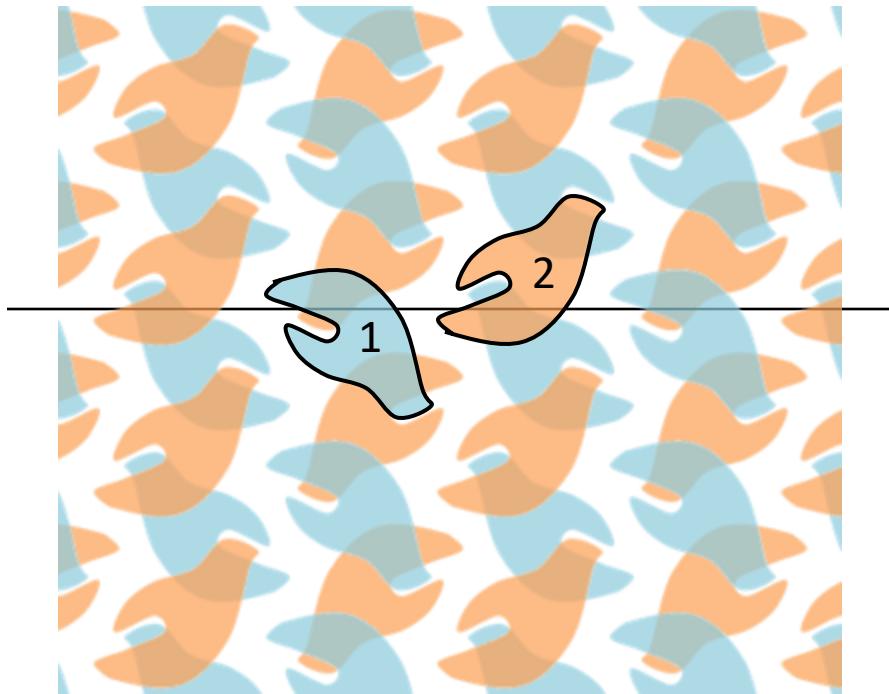


Can be fully characterised by six numbers  
(the third dimension is not shown here)

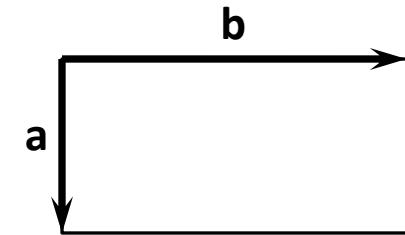
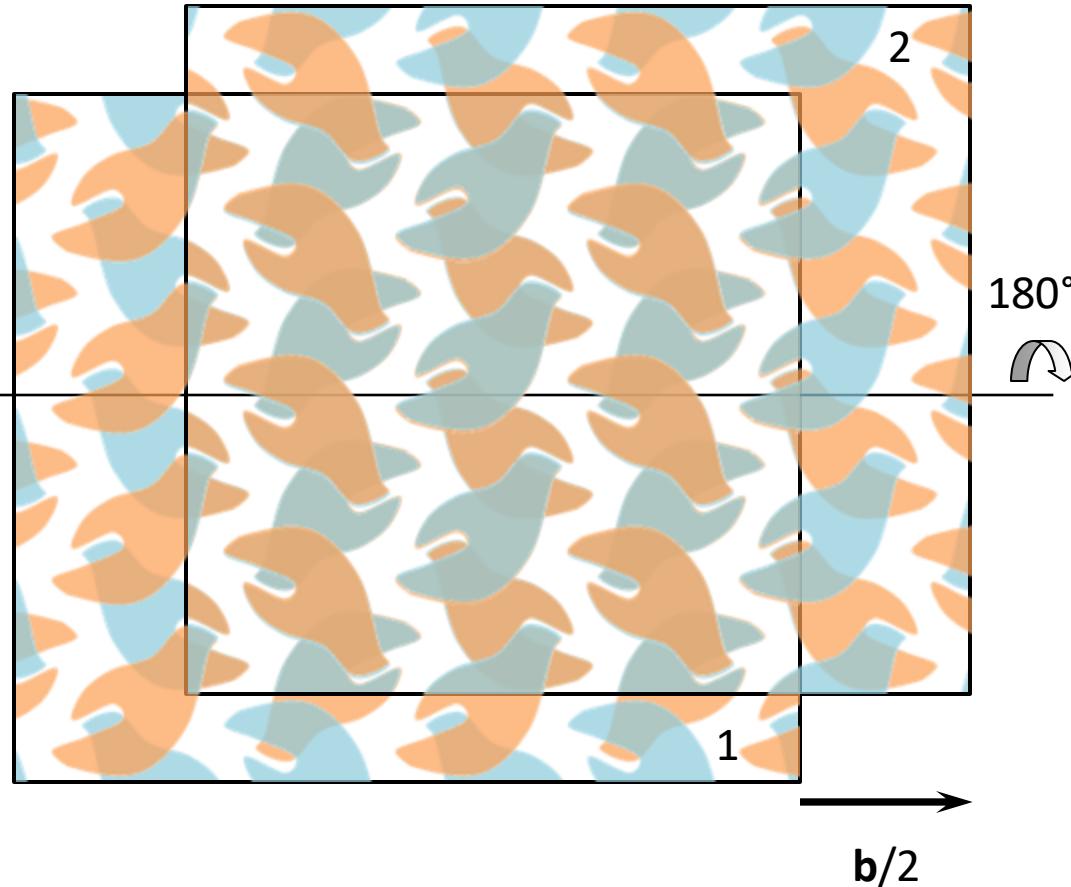
# Back to example



# Screw rotation



# Screw rotation axis



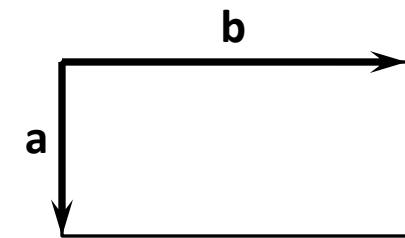
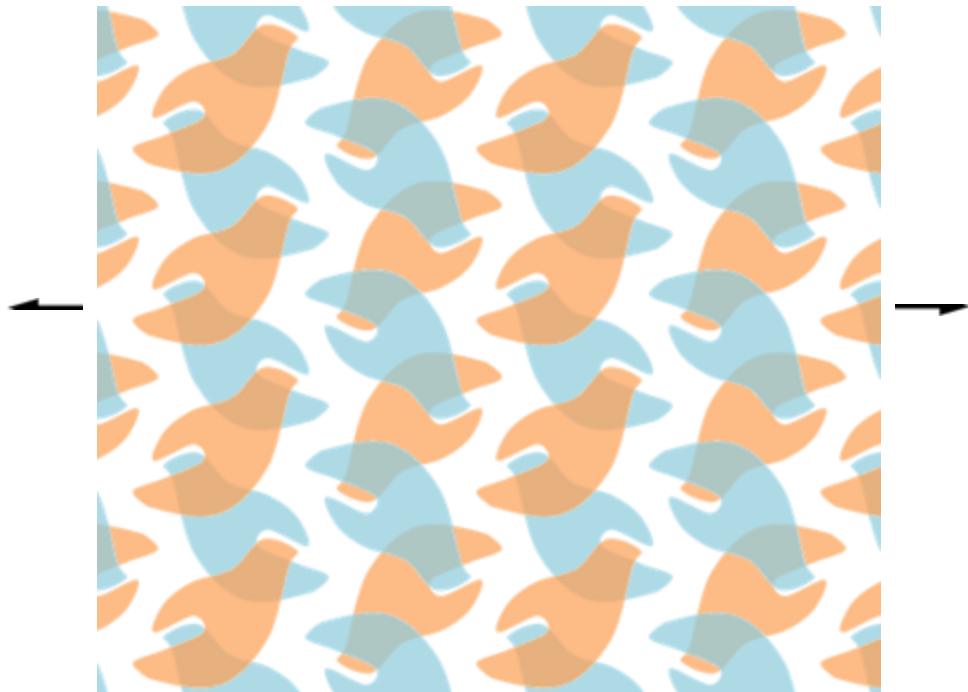
Operation  $1 \rightarrow 2$   
maps the whole crystal  
onto itself:

this is a **crystallographic  
operation**

The axis is a  
crystallographic  
symmetry element,

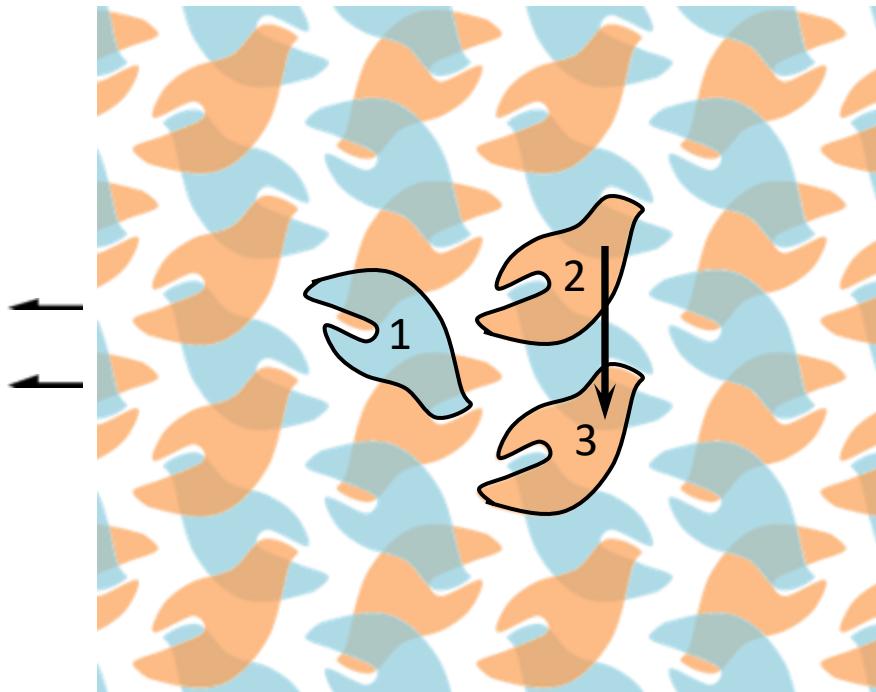
it can be **mapped into  
the structure**

# Screw rotation - symbol

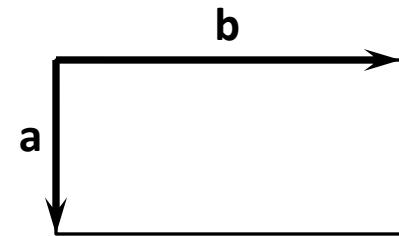


$2_1$  (plane of figure):

# Screw rotation - repeats



→ 1 -> 2  
→ 1 -> 3

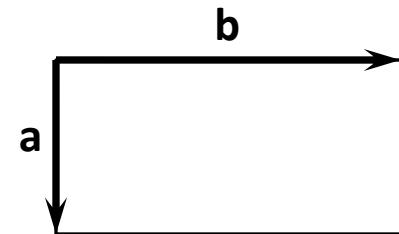
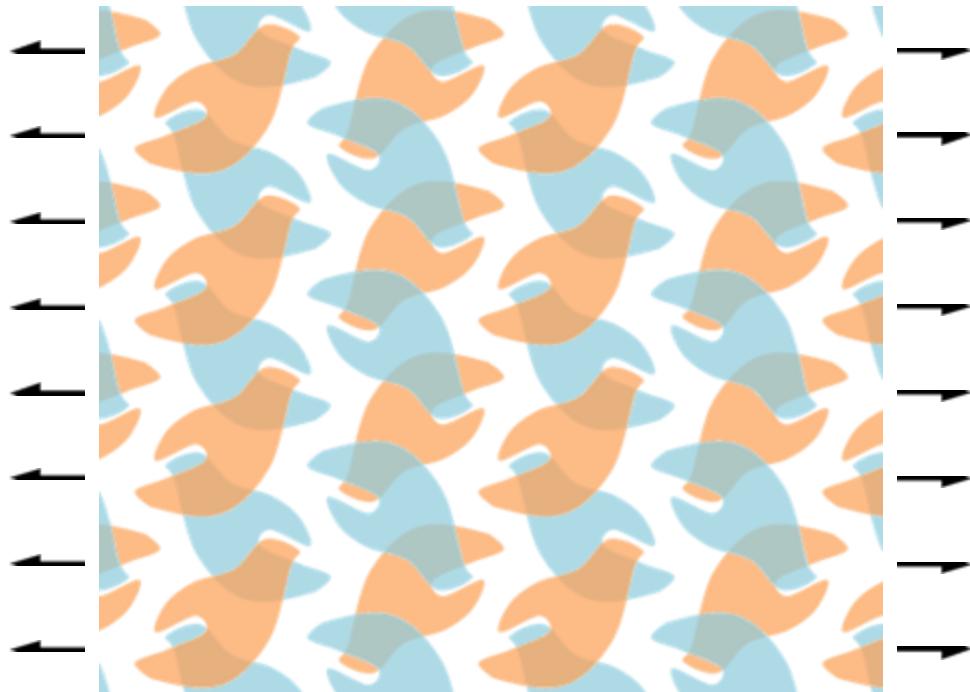


action of top axis  
×  
translation **a**  
=

action of bottom axis

(elements of a group)

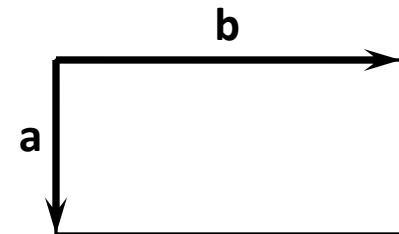
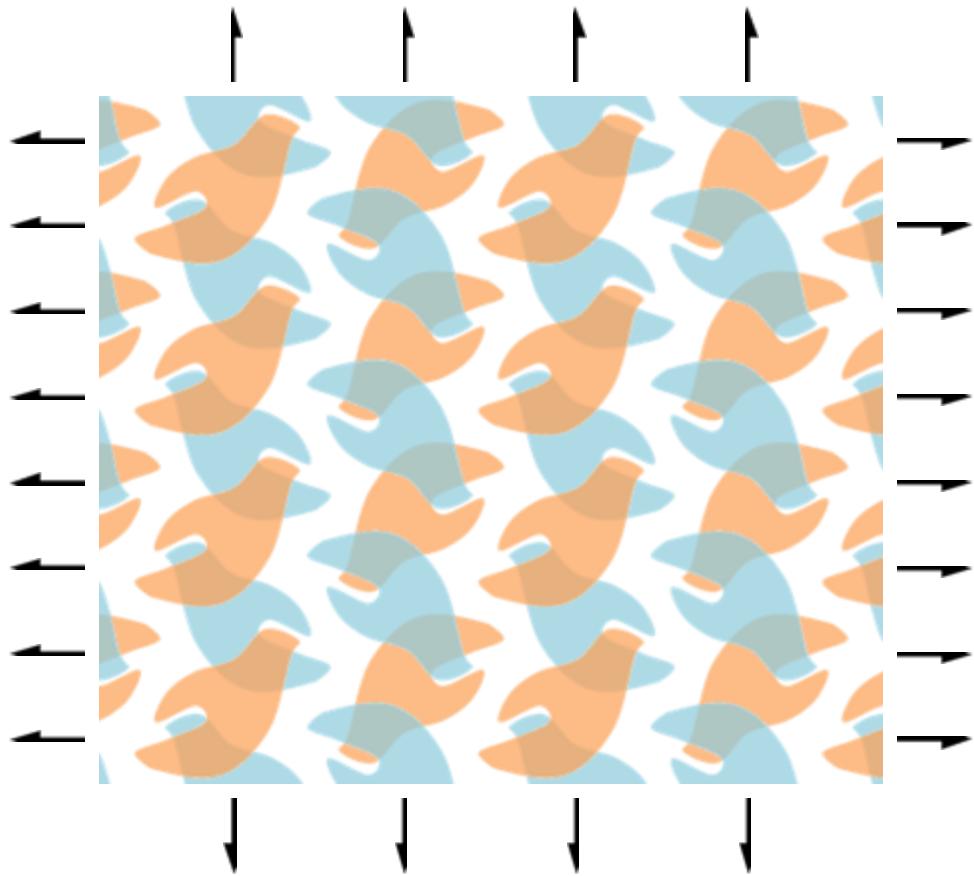
# Screw rotation 1 - repeats



$2_1$  (plane of figure):

Also repeated in 3-dimension  
with offset of  $\frac{1}{2} c$

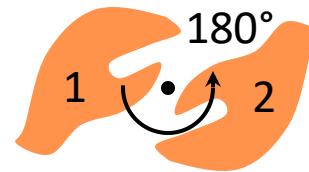
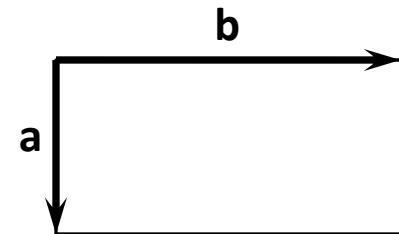
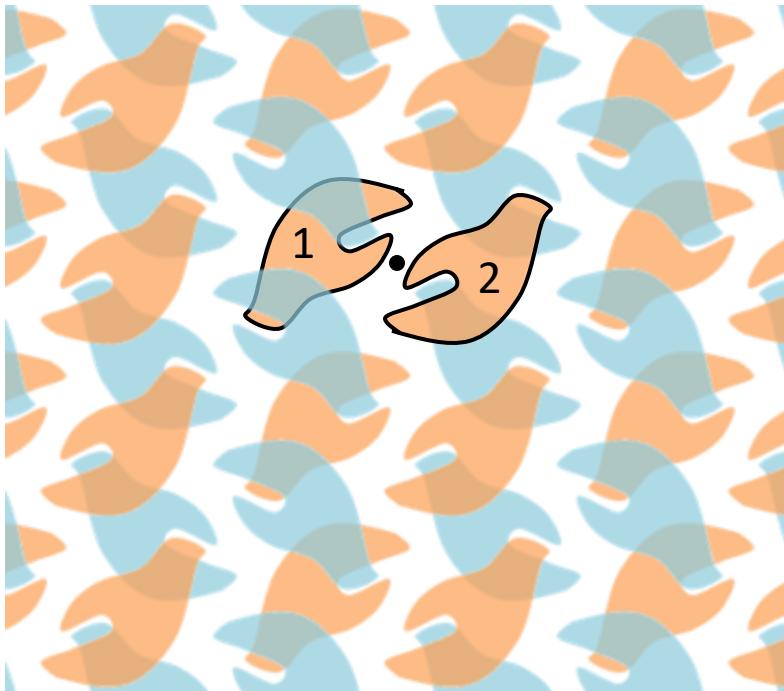
# Screw rotations parallel to **a** and **b**



$2_1$  (plane of figure): ← →

Series of  $2_1$  axes offset by  
 $\frac{1}{2}$  unit cell from each other.

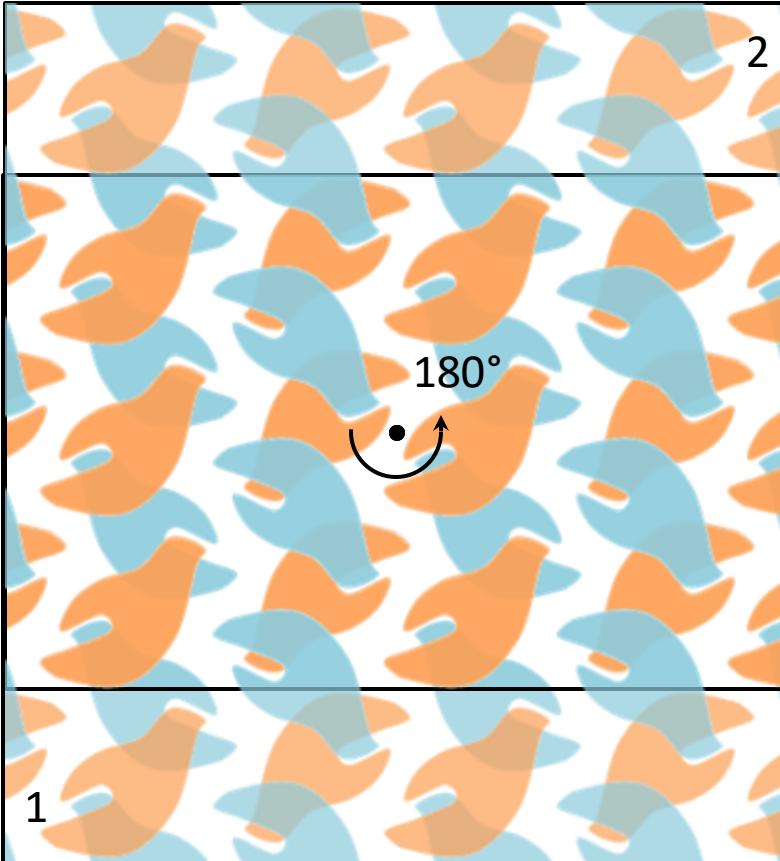
# Screw rotation – into plane



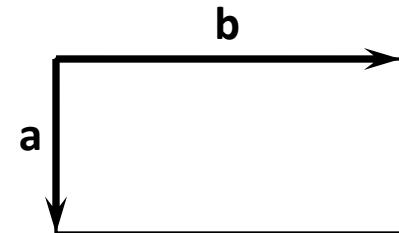
$\odot$   
 $c/2$

A rotation of 180° with a translation  
of  $\frac{1}{2}$  unit cell from the figure.

# Screw rotation 3 is global



⊕  
 $c/2$



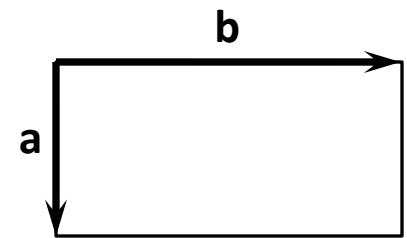
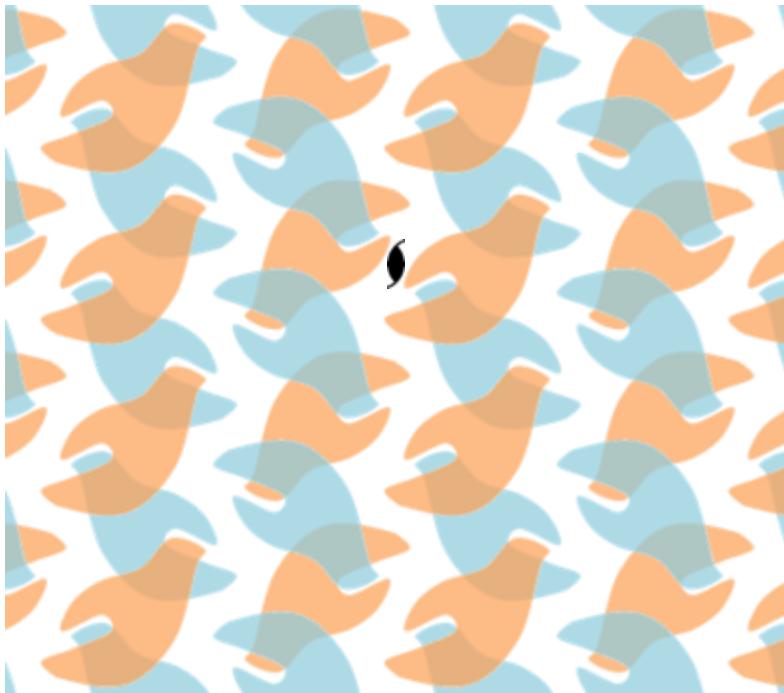
Screw rotation 3  
maps the whole crystal  
onto itself:

this is a **crystallographic operation**

The rotation axis is a  
crystallographic  
symmetry element,

it can be **mapped into  
the structure**

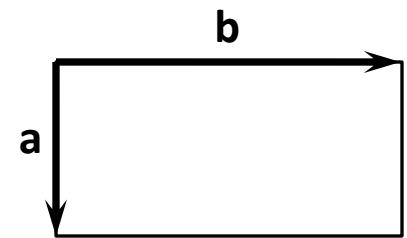
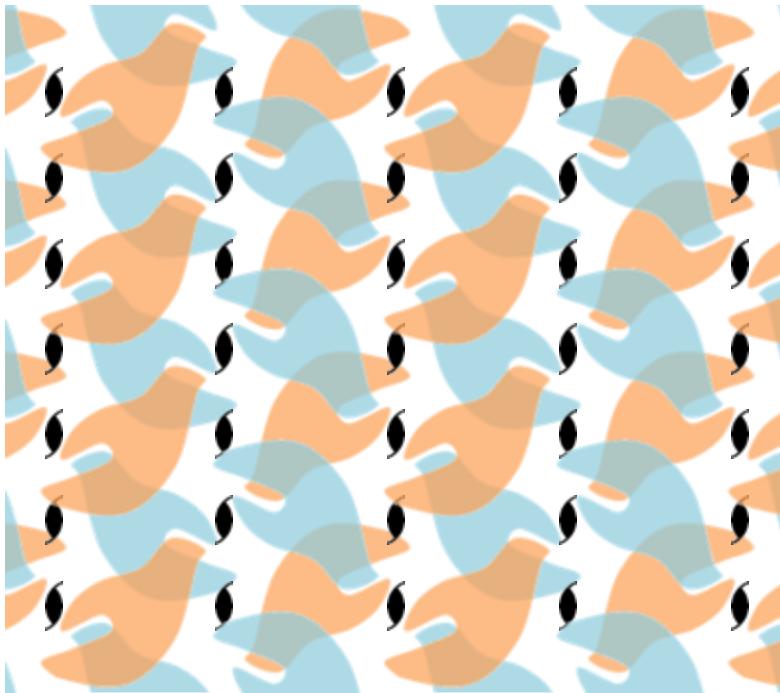
# Screw rotation 3 - symbol



$2_1$  (along view):



# Screw rotation 3 - repeats

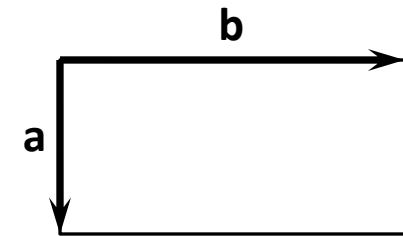
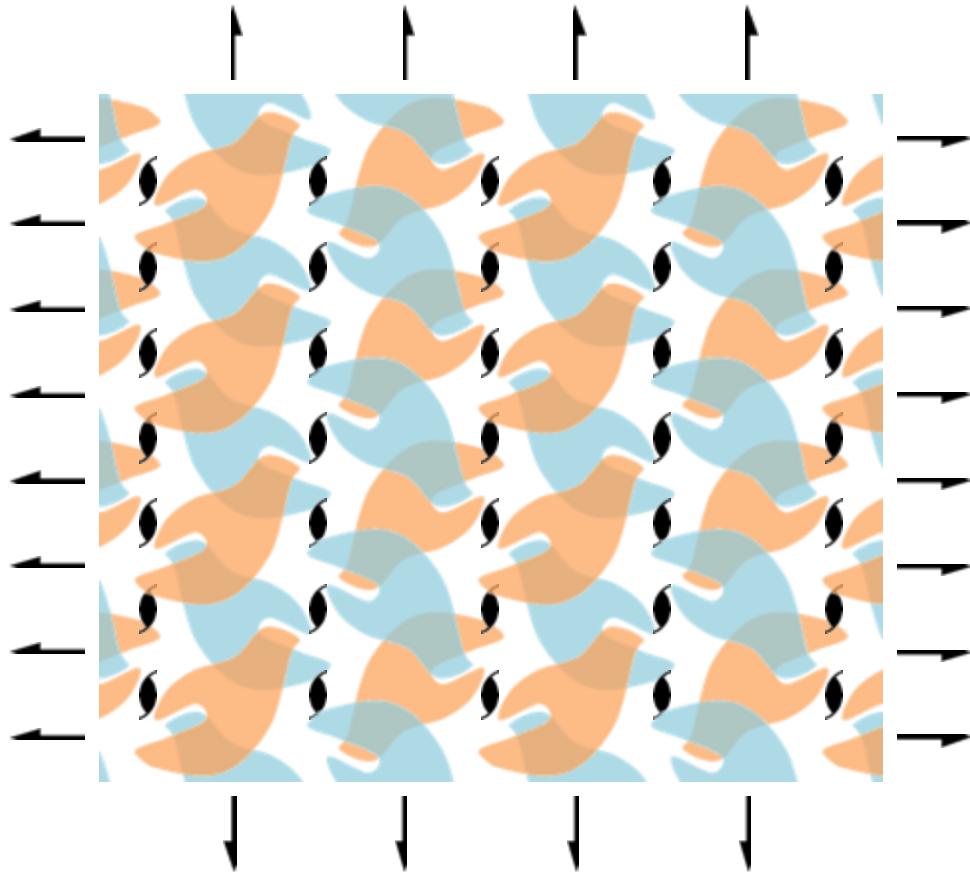


$2_1$  (along view):



As for the in-plane axes,  
there are repeated axes  
into the plane

# All axes together



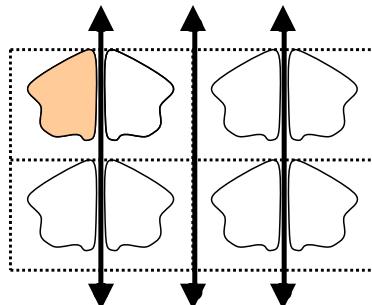
$2_1$  (plane of figure):

$2_1$  (along view):

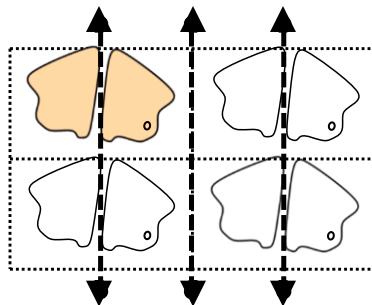
we have built a  
**space group**

- identity
- inverse
- multiplication

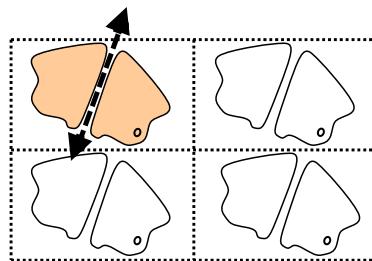
# Crystallographic Symmetry, Pseudosymmetry and Non-Crystallographic Symmetry (NCS)



Crystallographic symmetry  
- symmetry is **global** and **exact**

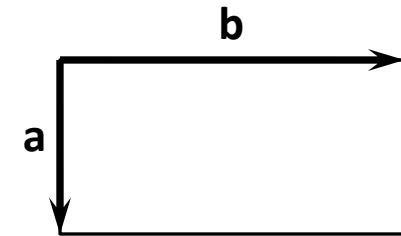
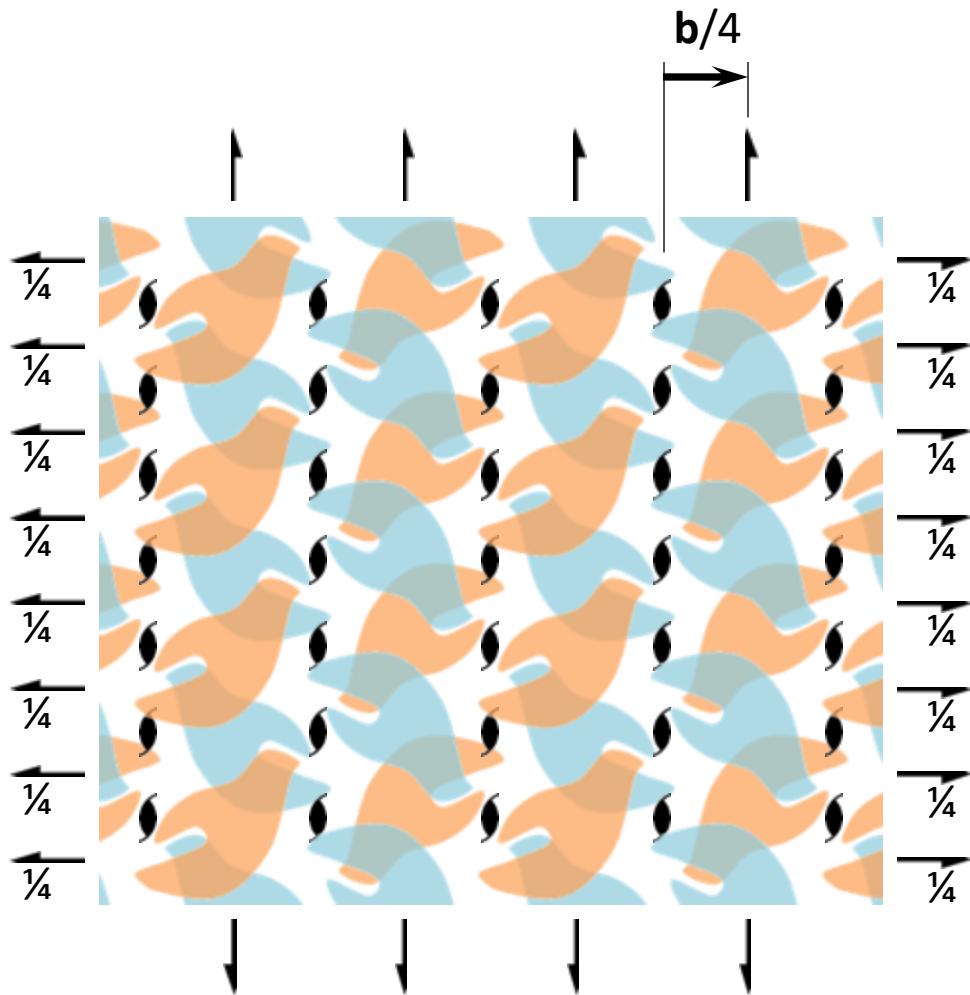


Pseudosymmetry (a limiting case of NCS)  
- symmetry is **global** and **approximate**



Generic Non-Crystallographic Symmetry (NCS):  
- symmetry is **local** and **approximate**

# Relative positions of axes



$2_1$  (plane of figure):

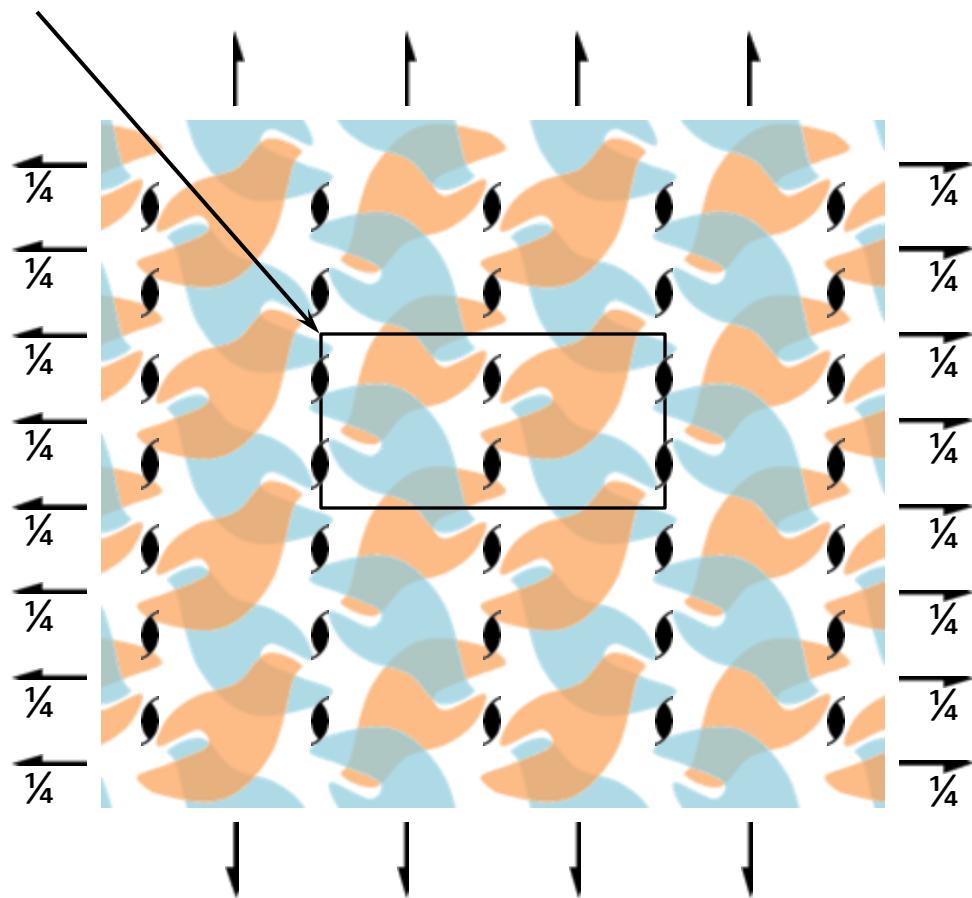
$2_1$  (along view):

The adjacent axes running in different directions are offset by  $\frac{1}{4}$  of corresponding base vector.

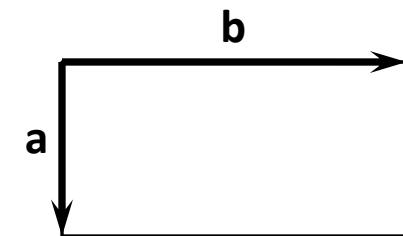
The horizontal  $\frac{1}{4}$  indicates a offset of  $(\frac{1}{2} n + \frac{1}{4}) c$  into the figure.

# Choice of origin is a convention. Notation

The origin ( $x=0, y=0, z=0$ )



is chosen to be equidistant from adjacent axes. Such a choice is a **convention**.



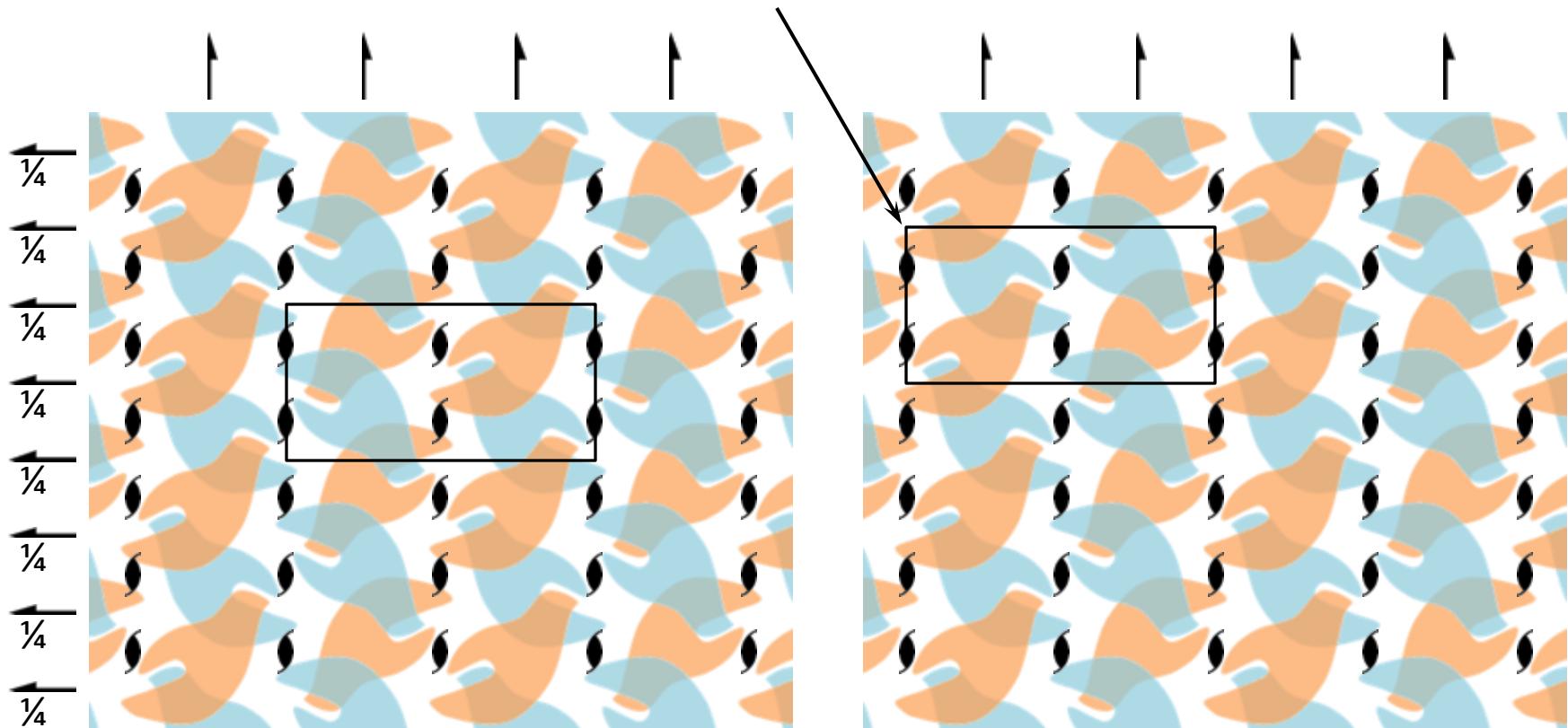
$2_1$  (plane of figure):

$2_1$  (along view):

The unit cell placed on picture with symmetry elements means a choice of origin.

# Alternative origins

Choose a different origin following the same convention (equidistant from adjacent axes)

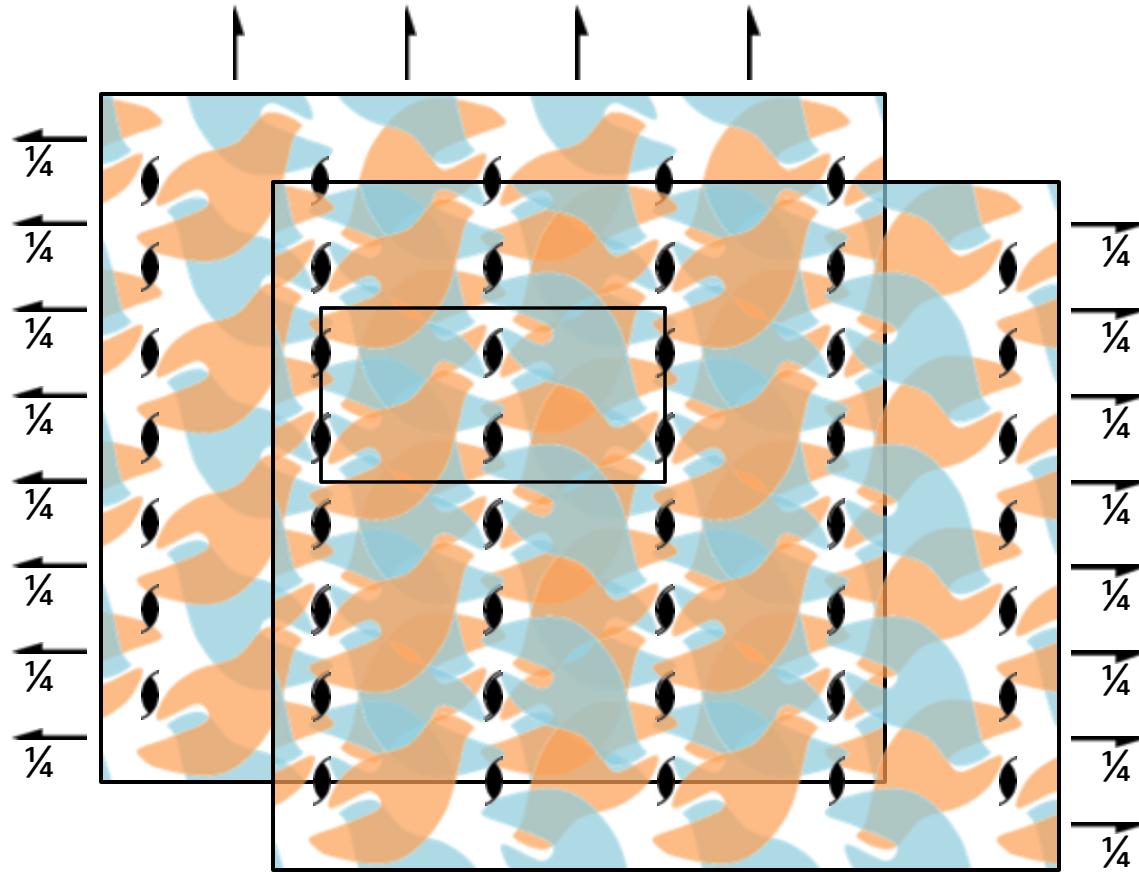


# Structures "solved in alternative origins"

Superpose two unit cells.

Positions of axes match.

The structures do not match.

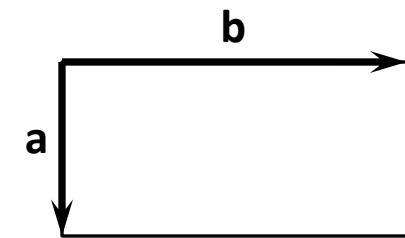
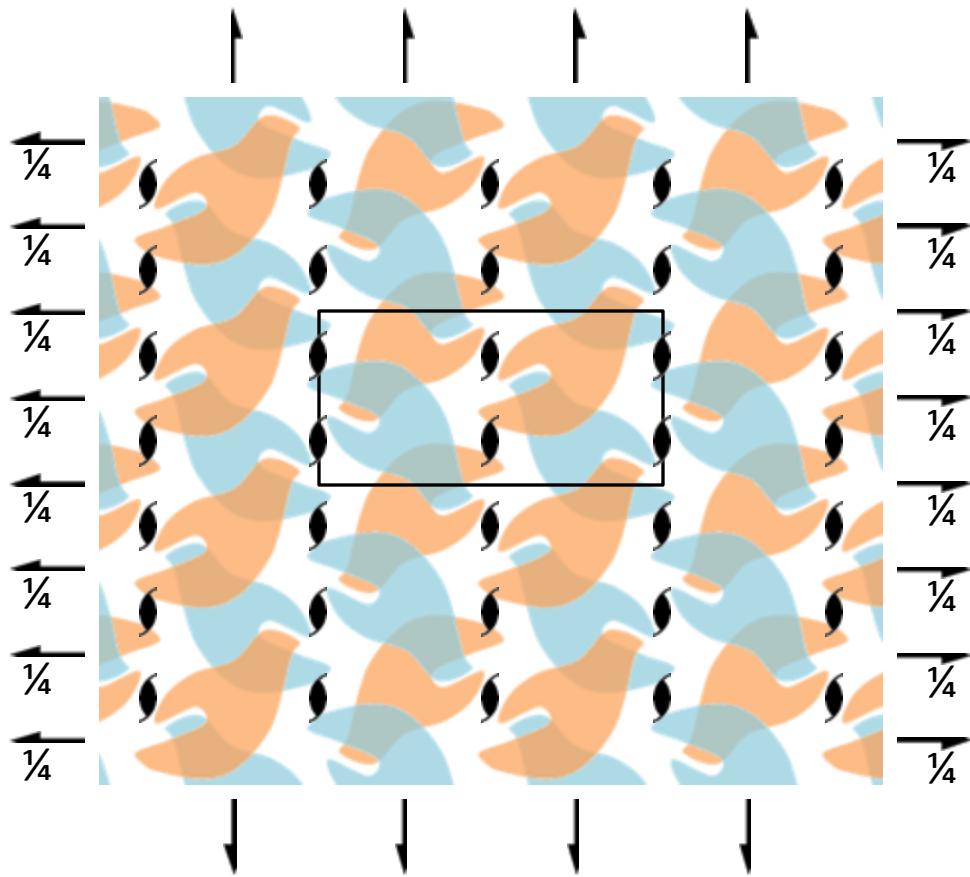


Conventions regarding the choice of crystallographic origin are expressed in terms of position relative to crystallographic axes.

Such a choice is not necessarily unique with respect to the structure.

Be conscious of "alternative" origins when e.g. comparing different MR solutions.

# Complete picture



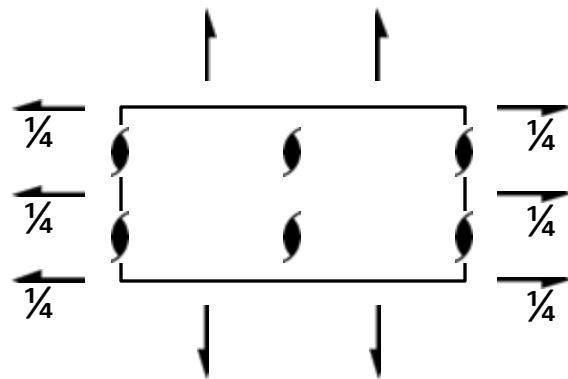
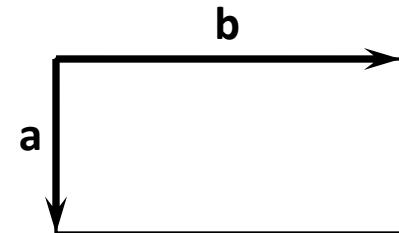
$2_1$  (plane of figure):

$2_1$  (along view):

# Compact representation

$P2_12_12_1$

No. 19



$2_1$  (plane of figure):

$2_1$  (along view):

Scheme with symmetry axes -> space group symbol -> more info in International Tables  
We will discuss space group symbols a bit later

# Space group representation in ITC-A

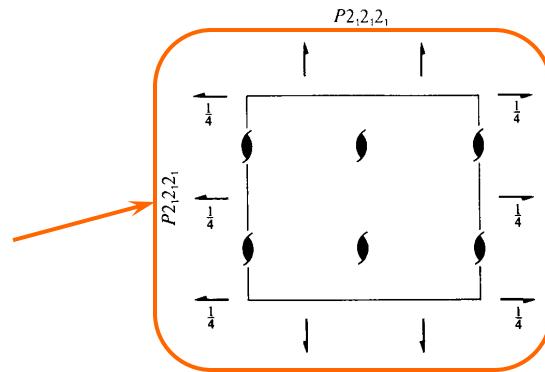
$P2_1 2_1 2_1$   
No. 19

$D_2^4$   
 $P2_1 2_1 2_1$

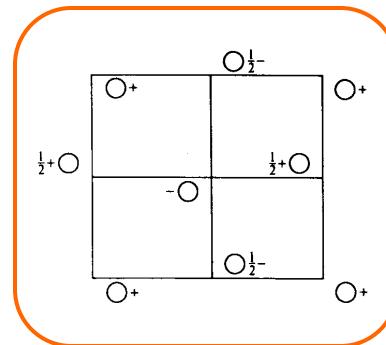
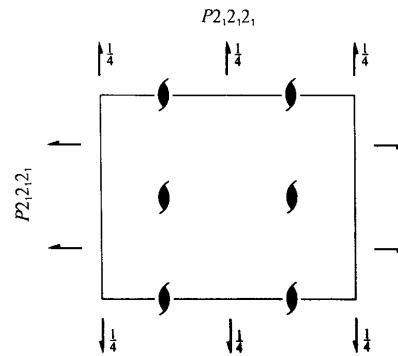
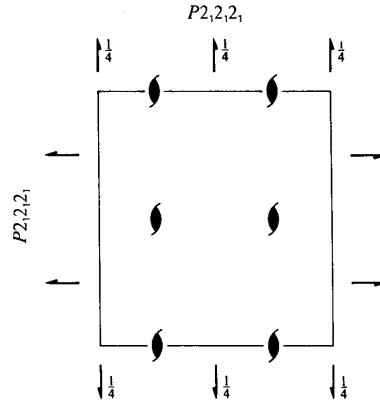
222

Orthorhombic  
Patterson symmetry  $Pmm$

Location of  
symmetry  
elements



Two other projections  
are also shown for this  
space group



Set of equivalent  
points in general  
position.

We will be looking at  
"molecular wallpaper"  
instead

# Space group representation in ITC-A

$P2_1 2_1 2_1$   
No. 19

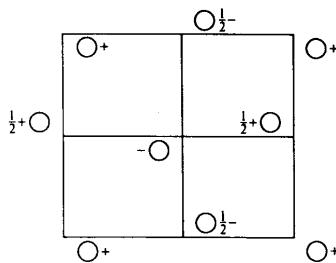
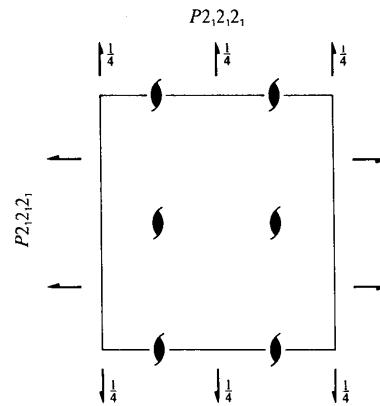
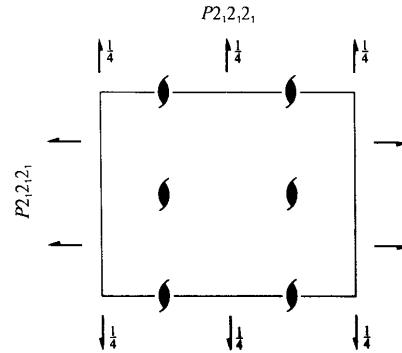
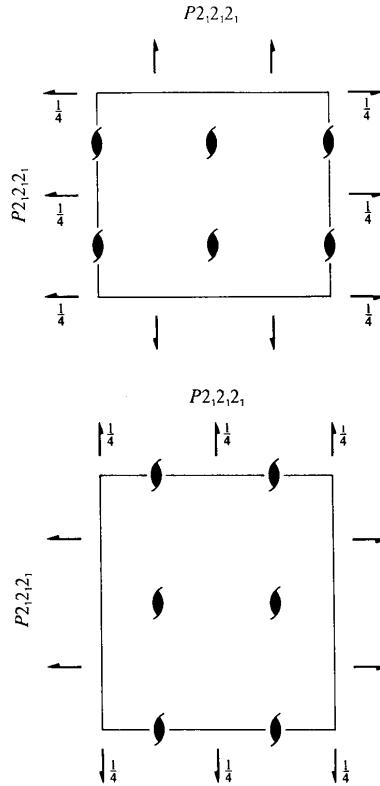
$D_2^4$   
 $P2_1 2_1 2_1$

222

Orthorhombic

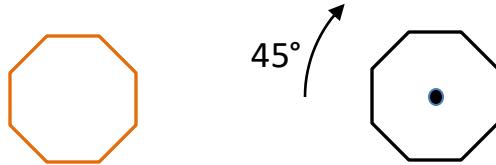
Patterson symmetry  $Pmm$

Crystal  
system



# Rotational symmetry

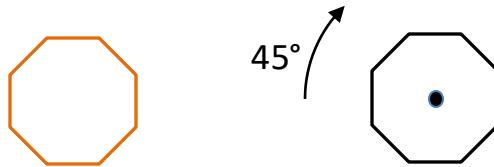
An **N**-fold rotational symmetry implies that if a rotation of  $360^\circ/\text{N}$  degrees is applied, the transformed object is identical to the original.



This object has **8**-fold rotational symmetry

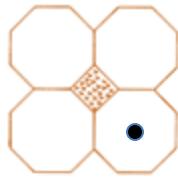
# Rotational symmetry

An **N**-fold rotational symmetry implies that if a rotation of  $360^\circ/\text{N}$  degrees is applied, the transformed object is identical to the original.

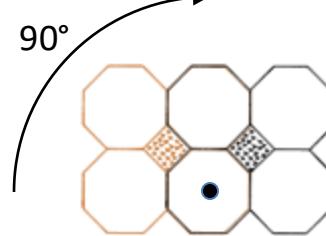


This object has **8-fold rotational symmetry**

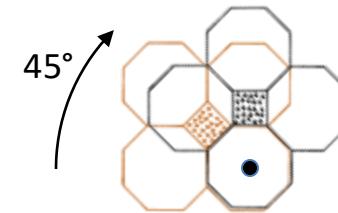
The only rotational symmetries possible in a **crystal** are **2, 3, 4** and **6**.



Wallpaper: objects with  
**8-fold symmetry**



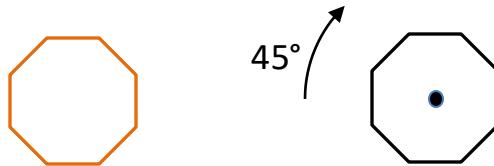
**4-fold rotational**  
**crystal** symmetry



**8-fold rotational**  
**crystal** symmetry  
is **impossible**

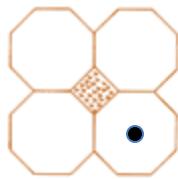
# Rotational symmetry

An **N**-fold rotational symmetry implies that if a rotation of  $360^\circ/\text{N}$  degrees is applied, the transformed object is identical to the original.

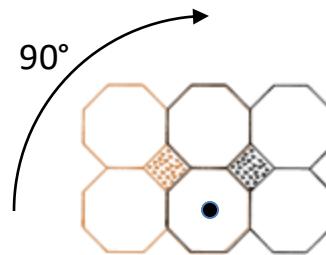


This object has **8-fold rotational symmetry**

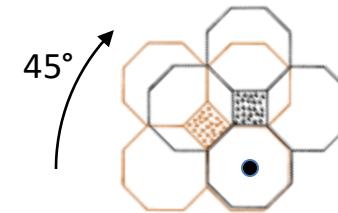
The only rotational symmetries possible in a **crystal** are **2, 3, 4** and **6**.



Wallpaper: objects with  
**8-fold symmetry**



**4-fold rotational**  
**crystal** symmetry

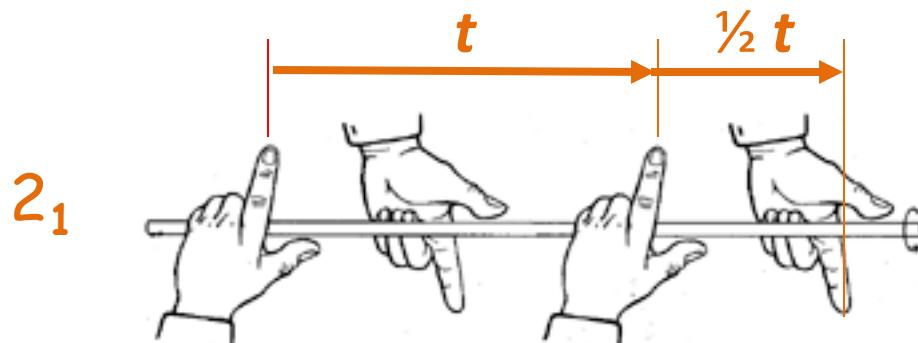


**8-fold rotational**  
**crystal** symmetry  
is **impossible**

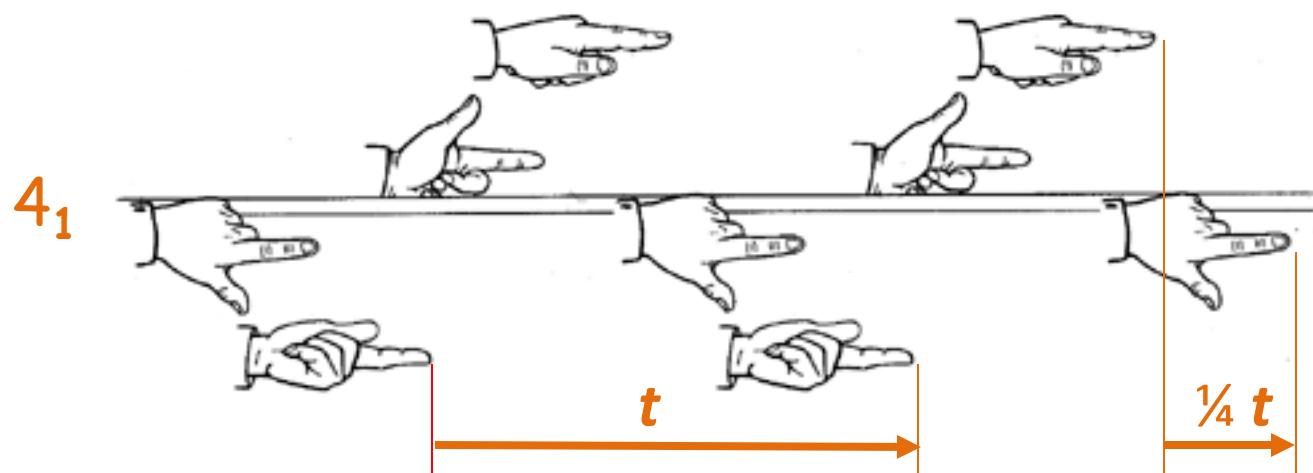
45° rotation is a **Non-Crystallographic Symmetry** (NCS)

# Screw axes

- Rotate clockwise about an axis (1, 2, 3, 4 or 6-fold rotation)
- Translate along this axis by a fraction of the shortest crystallographic translation along the rotation axis



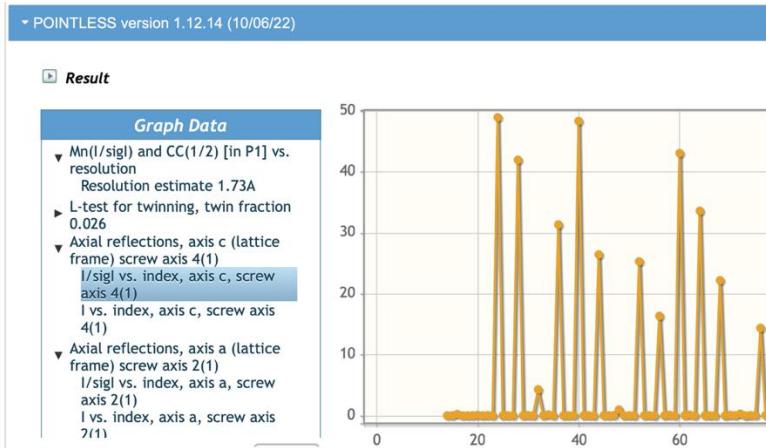
translation of  $\frac{1}{2}t$   
per 180° rotation



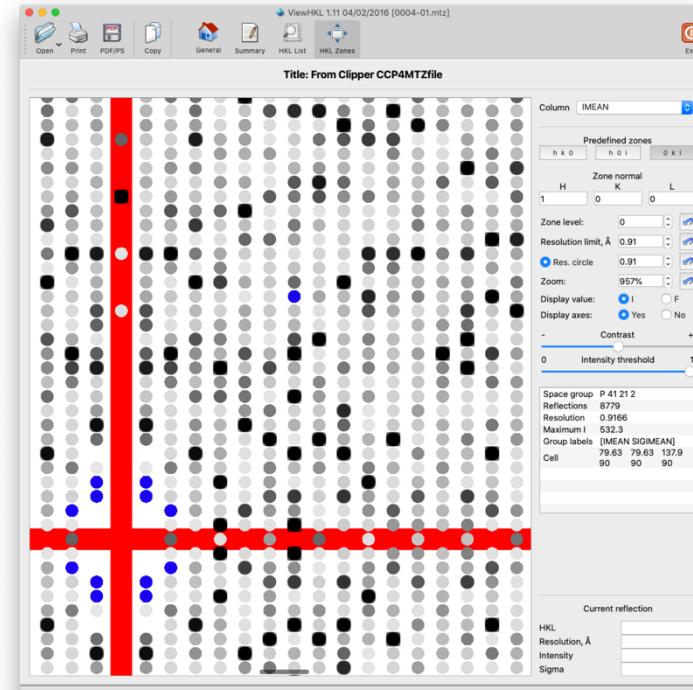
translation of  $\frac{1}{4}t$   
per 90° rotation

# Screw axes: systematic absences

$h = k = 0$ : every fourth  $l$  strong:  $4_1$  or  $4_3$

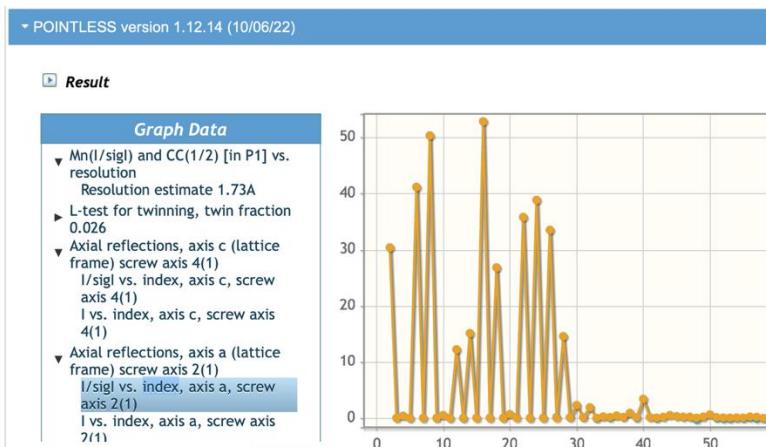


$4_1$  or  $4_3$



$2_1$

$k = l = 0$ : every second  $h$  strong:  $2_1$



$h$

$l$

- might not be very clear with noisy data
- enantiomorph not resolved:  $P4_12_12$  or  $P4_32_12$

# Screw axes: systematic absences: CCP4Cloud

[0001] file import -- imported: Unmerged (1)  
[0004] aimless -- Compl=83.0% CC<sub>1/2</sub>=0.999 R<sub>meas\_all</sub>=0.258 R<sub>meas\_ano</sub>=0.257 Res=0.91-39.81 SpG=P 41 21 2  
[0005] asymmetric unit contents  
[0006] file import -- imported: Sequence (1)

**[0004] aimless -- completed**

Input  Output + New Import Export

**Report Main Log Service Log Errors**

**3. Generating symmetry tables**

▼ POINTLESS version 1.12.14 (10/06/22)

▶ **Result**

**Graph Data**

▼ Mn(I/sigI) and CC(1/2) [in P1] vs. resolution  
Resolution estimate 1.73A

► L-test for twinning, twin fraction 0.026

► Axial reflections, axis c (lattice frame) screw axis 4(1)

18  
16  
14  
12  
10

**[0004] aimless -- completed**

Input  Output + New Import Export

**Report Main Log Service Log Errors**

**Created Reflection Data Set (merged)**

Assigned name: [0004-01] aimless [XDSproject/]

▶ **Reflection data** ViewHKL

**[0004] References**

The following programs were used:

- **Pointless:**
  - Evans, P.R. (2006) *Scaling and assessment*  
[doi:10.1107/S0907444905036693](https://doi.org/10.1107/S0907444905036693)

# Symmetry elements allowed in chiral structures

Apart from the identity and translations, **macromolecular crystals** can only contain the following symmetry elements:

Proper Rotations	Screw Rotations

# Symmetry elements disallowed by chiral centres

Small molecules also face other symmetry operations

- Mirror plane **m**
- Glide planes **a**, **b**, **c**, **n** or **d**: reflection across plane followed by translation parallel to plane along **a**, **b**, **c**, **face diagonal** or **body diagonal**, respectively
- Rotation – inversion  $\bar{1}, \bar{3}, \bar{4}, \bar{6}$ : a rotation  $\bar{1}$  followed by inversion

# Space groups

- All possible combinations of symmetry elements => 230 space groups
- Because protein and nucleic acid molecules are chiral, there are only 65 “biological” space groups.
- Space groups are divided on 7 crystal system based on
  - the presence of symmetry elements of a certain order (6, 4, 3, 2)
  - the number of different orientations of these elements

# Crystal Systems

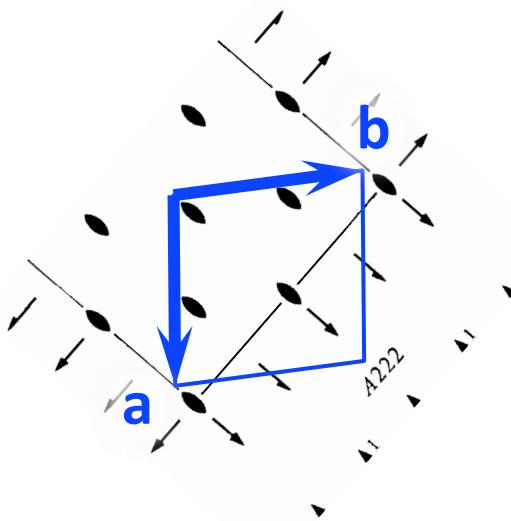
Crystal System	Characteristic symmetry elements	Convention	Bravais lattices	Constraints on unit cell parameters
1. Triclinic	Translations only		1. Primitive ( <i>P</i> )	
2. Monoclinic	2-fold axes, all parallel	along <b>b</b>	2. Primitive ( <i>P</i> ) 3. Base-Centered ( <i>C</i> )	$\alpha = \gamma = 90^\circ$
3. Orthorhombic	2-fold axes in three perpendicular directions  <b>(example)</b>	along <b>a</b> , <b>b</b> and <b>c</b>	4. Primitive ( <i>P</i> ) 5. Base-Centered ( <i>C</i> ) 6. Body-Centered ( <i>I</i> ) 7. Face-Centered ( <i>F</i> )	$\alpha = \beta = \gamma = 90^\circ$
4. Tetragonal	4-fold axes, all parallel	along <b>c</b>	8. Primitive ( <i>P</i> ) 9. Body-Centered ( <i>I</i> )	$a = b$ $\alpha = \beta = \gamma = 90^\circ$
5. Trigonal	3-fold axes, all parallel	along <b>c</b>	10. Primitive ( <i>P</i> ) 11. Rhombohedral ( <i>R / H</i> )	$a = b$ $\alpha = \beta = 90^\circ; \gamma = 120^\circ$
6. Hexagonal	6-fold axes, all parallel	along <b>c</b>	10. Primitive ( <i>P</i> )	$a = b$ $\alpha = \beta = 90^\circ; \gamma = 120^\circ$
7. Cubic	3-fold axes in four different orientations	along body diagonals	12. Primitive ( <i>P</i> ) 13. Body-Centered ( <i>I</i> ) 14. Face-Centered ( <i>F</i> )	$a = b = c$ $\alpha = \beta = \gamma = 90^\circ$

# Crystal Systems

Crystal System	Characteristic symmetry elements	Convention	Bravais lattices	Constraints on unit cell parameters
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2. Monoclinic	2-fold axes, all parallel	along <b>b</b>	2. Primitive ( <i>P</i> ) 3. Base-Centered ( <i>C</i> )	$\alpha = \gamma = 90^\circ$
3. Orthorhombic	2-fold axes in three perpendicular directions	along <b>a</b> , <b>b</b> and <b>c</b>	4. Primitive ( <i>P</i> ) 5. Base-Centered ( <i>C</i> ) 6. Body-Centered ( <i>I</i> ) 7. Face-Centered ( <i>F</i> )	$\alpha = \beta = \gamma = 90^\circ$
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6. Hexagonal	6-fold axes, all parallel	along <b>c</b>	10. Primitive ( <i>P</i> )	$a = b$ $\alpha = \beta = 90^\circ; \gamma = 120^\circ$
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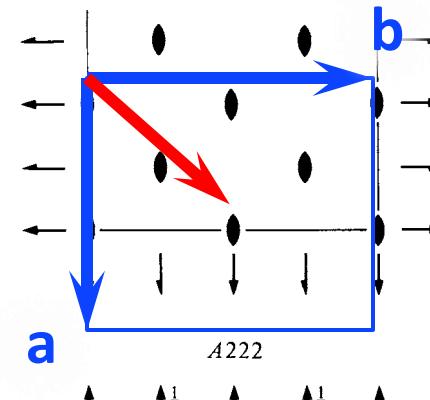
# C222: an example of a centred cell

If we were using  
a primitive cell



2-fold axes are along  
face diagonals  
(non-conventional  
crystal setting)

Standard setting;  
C means additional  
translation  $\frac{1}{2} (\mathbf{a} + \mathbf{b})$

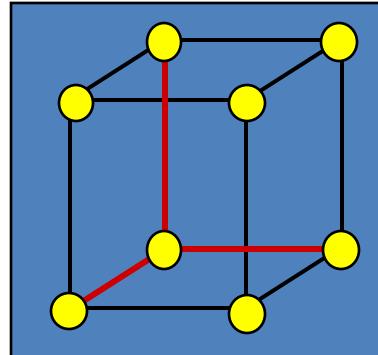
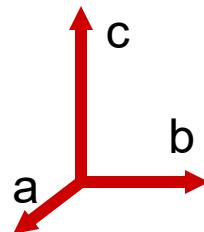


2-fold axes are  
along **a**, **b** and **c**  
(conventional  
setting)

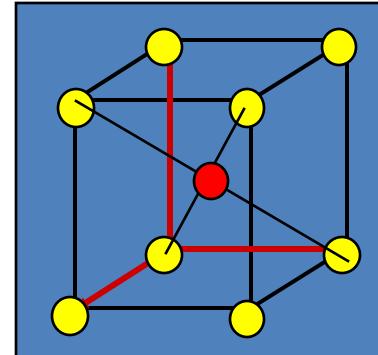
# Centred unit cells in pictures

Our convention dictates that direction(s) of rotation axes define **a**, **b** and **c**

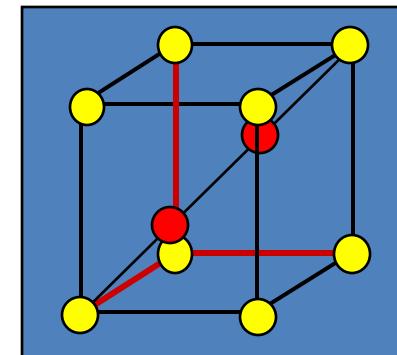
As a result, the crystal lattice of some space groups contains "additional" nodes (red) that represent "additional" translations



P – Primitive

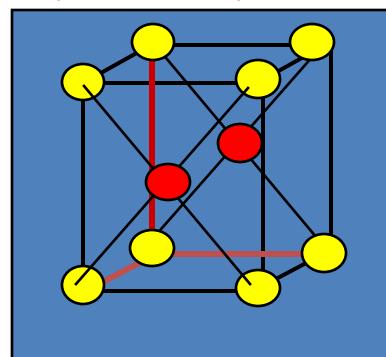


I – Body centred



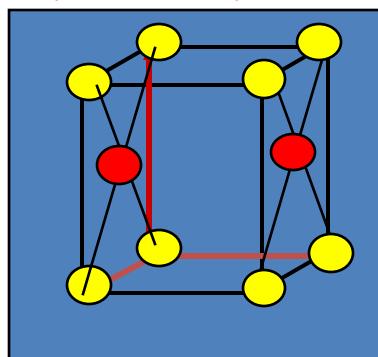
H – Hexagonal

(non-standard)

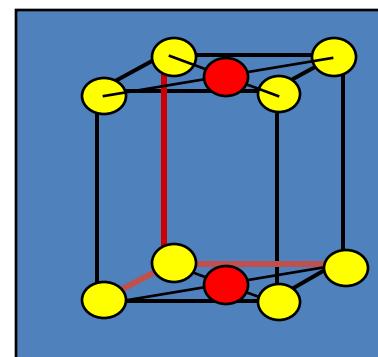


A – Face centred (A)

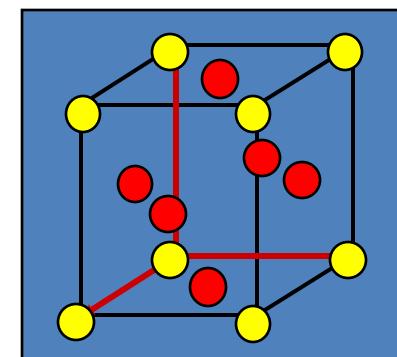
(non-standard)



B – Face centred (B)



C – Face centred (C)

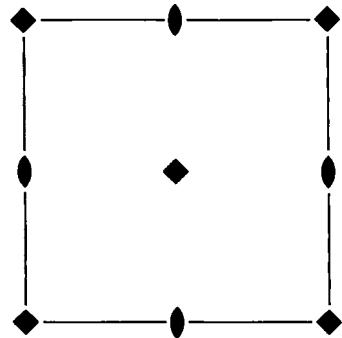


F – Face centred (all)

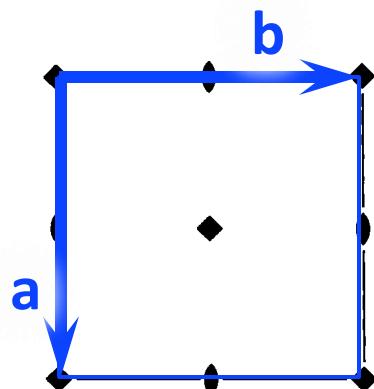
# C4: an example of a redundant space group symbol

P4

as presented in the International Tables for Crystallography

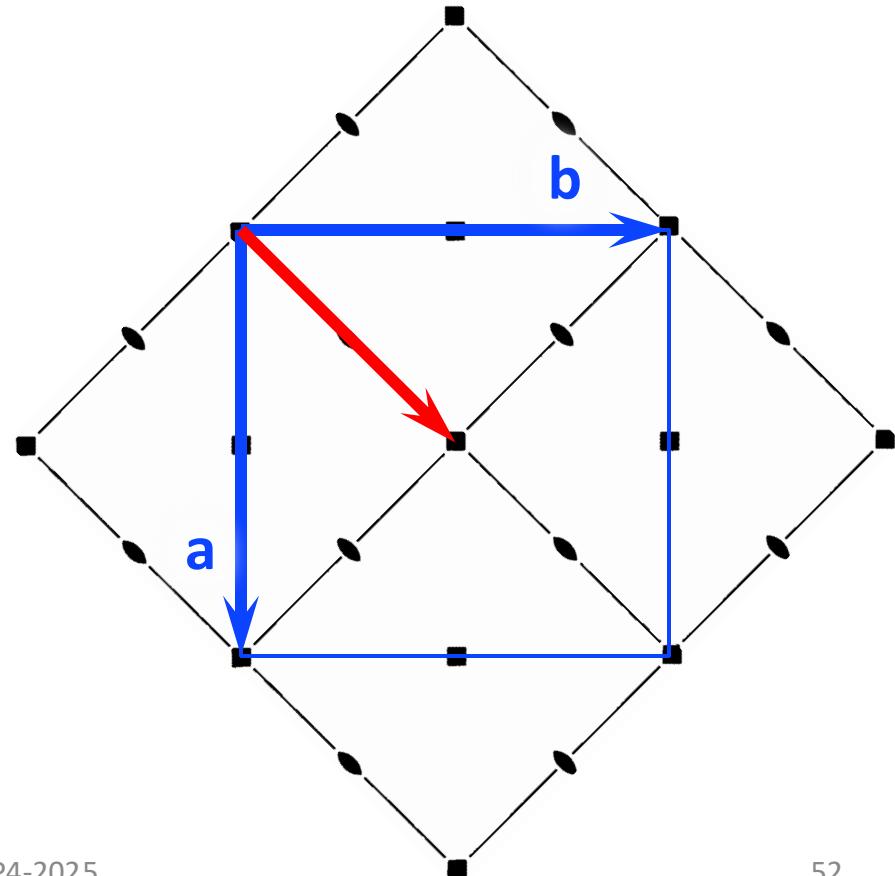


and with base vectors shown



C4

This is just a different setting obtained  
- by rotation 45° and  
- redefining base vectors  
- additional translation  $(\mathbf{a} + \mathbf{b})/ 2$



# Bravais lattices

- 7 crystal systems, combined with some of the centring types (P, C, I, F or H) give 14 Bravais lattices
- Not included:
  - impossible combinations (*e.g.* A4)
  - redundant combinations (*e.g.* P4 is kept, C4 is excluded)

# Crystal Systems

Crystal System	Characteristic symmetry elements	Convention	Bravais lattices	Constraints on unit cell parameters
1. Triclinic	Translations only		1. Primitive ( <i>P</i> )	
2. Monoclinic	2-fold axes, all parallel	along <b>b</b>	2. Primitive ( <i>P</i> ) 3. Base-Centered ( <i>C</i> )	$\alpha = \gamma = 90^\circ$
3. Orthorhombic	2-fold axes in three perpendicular directions	along <b>a</b> , <b>b</b> and <b>c</b>	4. Primitive ( <i>P</i> ) 5. Base-Centered ( <i>C</i> ) 6. Body-Centered ( <i>I</i> ) 7. Face-Centered ( <i>F</i> )	$\alpha = \beta = \gamma = 90^\circ$
4. Tetragonal	4-fold axes, all parallel	along <b>c</b>	8. Primitive ( <i>P</i> ) 9. Body-Centered ( <i>I</i> )	$a = b$ $\alpha = \beta = \gamma = 90^\circ$
5. Trigonal	3-fold axes, all parallel	along <b>c</b>	10. Primitive ( <i>P</i> ) 11. Rhombohedral ( <i>R / H</i> )	$a = b$ $\alpha = \beta = 90^\circ; \gamma = 120^\circ$
6. Hexagonal	6-fold axes, all parallel	along <b>c</b>	10. Primitive ( <i>P</i> )	$a = b$ $\alpha = \beta = 90^\circ; \gamma = 120^\circ$
7. Cubic	3-fold axes in four different orientations	along body diagonals	12. Primitive ( <i>P</i> ) 13. Body-Centered ( <i>I</i> ) 14. Face-Centered ( <i>F</i> )	$a = b = c$ $\alpha = \beta = \gamma = 90^\circ$

# Crystal Systems

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3. Orthorhombic	2-fold axes in three perpendicular directions	along <b>a</b> , <b>b</b> and <b>c</b>	4. Primitive ( <i>P</i> ) 5. Base-Centered ( <i>C</i> ) 6. Body-Centered ( <i>I</i> ) 7. Face-Centered ( <i>F</i> )	$\alpha = \beta = \gamma = 90^\circ$
4. Tetragonal	4-fold axes, all parallel	along <b>c</b>	8. Primitive ( <i>P</i> ) 9. Body-Centered ( <i>I</i> )	$a = b$ $\alpha = \beta = \gamma = 90^\circ$
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# Crystal Systems

Crystal System	Characteristic symmetry elements	Convention	Bravais lattices	Constraints on unit cell parameters
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2. Monoclinic	2-fold axes, all parallel	along <b>b</b>	2. Primitive ( <i>P</i> ) 3. Base-Centered ( <i>C</i> )	$\alpha = \gamma = 90^\circ$
3. Orthorhombic	2-fold axes in three perpendicular directions	along <b>a</b> , <b>b</b> and <b>c</b>	4. Primitive ( <i>P</i> ) 5. Base-Centered ( <i>C</i> ) 6. Body-Centered ( <i>I</i> ) 7. Face-Centered ( <i>F</i> )	$\alpha = \beta = \gamma = 90^\circ$
4. Tetragonal	4-fold axes, all parallel	along <b>c</b>	8. Primitive ( <i>P</i> ) 9. Body-Centered ( <i>I</i> )	$a = b$ $\alpha = \beta = \gamma = 90^\circ$
5. Trigonal	3-fold axes, all parallel	along <b>c</b>	10. Primitive ( <i>P</i> ) 11. Rhombohedral ( <i>R / H</i> )	$a = b$ $\alpha = \beta = 90^\circ; \gamma = 120^\circ$
6. Hexagonal	6-fold axes, all parallel	along <b>c</b>	10. Primitive ( <i>P</i> )	$a = b$ $\alpha = \beta = 90^\circ; \gamma = 120^\circ$
7. Cubic	3-fold axes in four different orientations	along body diagonals	12. Primitive ( <i>P</i> ) 13. Body-Centered ( <i>I</i> ) 14. Face-Centered ( <i>F</i> )	$a = b = c$ $\alpha = \beta = \gamma = 90^\circ$

# Space group representation in ITC-A

(Short) Hermann-Mauguin symbol

$P2_1 2_1 2_1$

No. 19

(Extended) Hermann-Mauguin symbol

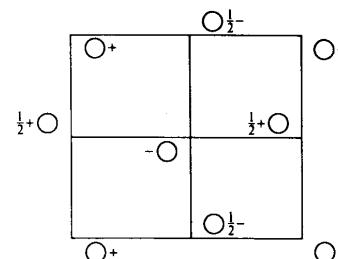
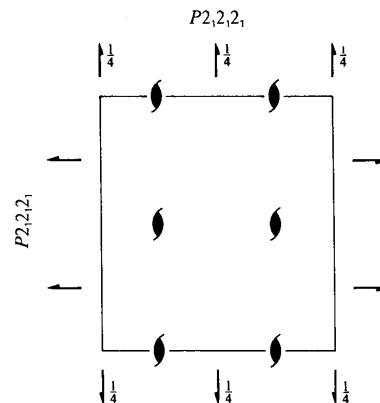
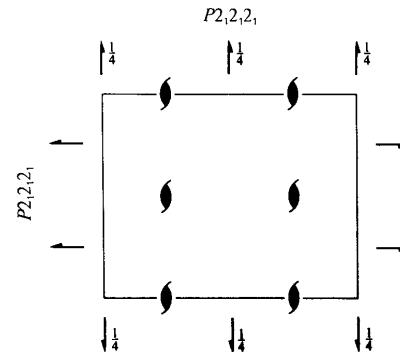
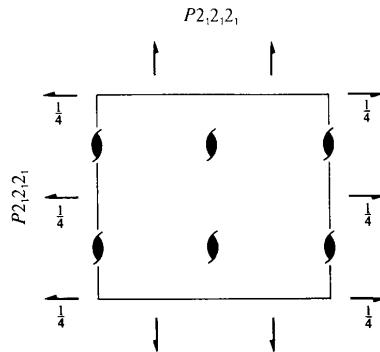
$D_2^4$

$P2_1 2_1 2_1$

222

Orthorhombic

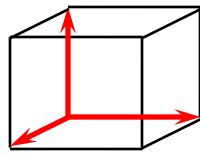
Patterson symmetry  $Pmm$



# Triclinic

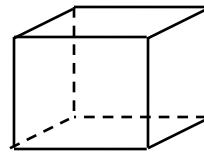
P 1

Lattice type



**P**

"1" means no symmetry operations except for translations



**1**

No constraints on  
 $a, b, c, \alpha, \beta, \gamma$

# Monoclinic

$P\bar{2}$   
 $C\bar{2}$

$P\bar{2}_1$

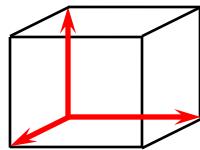
Standard HM symbols: **papers**

$P\bar{1}\bar{2}\bar{1}$   
 $C\bar{1}\bar{2}\bar{1}$

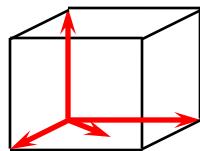
$P\bar{1}\bar{2}_1\bar{1}$

Extended HM symbols: **PDB**

Lattice type



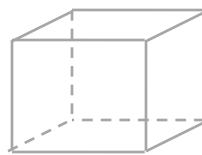
or



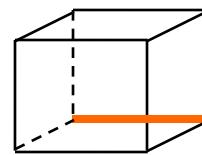
**P**

**C**

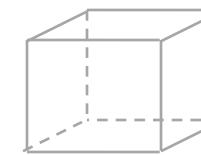
Directions and orders of axes



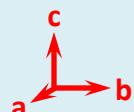
**1**



**2<sub>x</sub>**



**1**



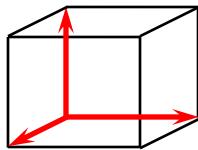
2-fold axes, all parallel

$\alpha = \gamma = 90^\circ$

# Orthorhombic

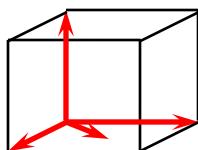
$P\bar{2}\bar{2}\bar{2}$     $P\bar{2}\bar{2}2_1$     $P2_12_12$     $P2_12_12_1$   
 $C\bar{2}\bar{2}\bar{2}$     $C\bar{2}\bar{2}2_1$   
 $I\bar{2}\bar{2}\bar{2}$     $I2_12_12_1$   
 $F\bar{2}\bar{2}\bar{2}$

Lattice type



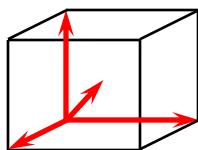
**P**

or



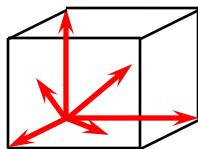
**C**

or



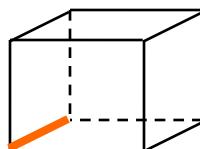
**I**

or

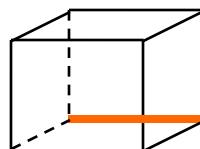


**F**

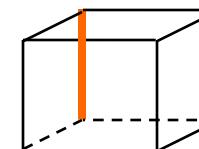
Directions and orders of axes



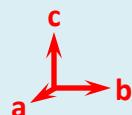
**2<sub>x</sub>**



**2<sub>x</sub>**



**2<sub>x</sub>**



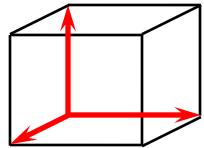
2-fold axes in three  
perpendicular directions

$\alpha = \beta = \gamma = 90^\circ$

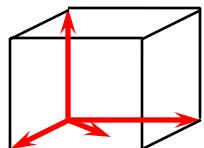
# Orthorhombic

$P\bar{2}\bar{2}\bar{2}$     $\bar{P}222_1$     $P2_12_12$     $P2_12_12_1$    example  
 $C2\bar{2}\bar{2}$     $C222_1$   
 $I\bar{2}\bar{2}\bar{2}$   
 $F2\bar{2}\bar{2}$

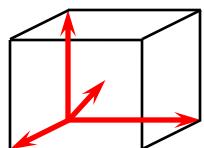
Lattice type



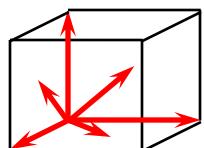
or



or



or



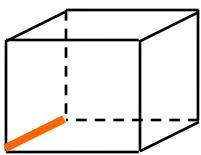
**P**

**C**

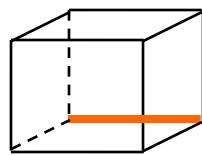
**I**

**F**

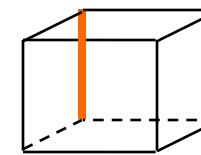
Directions and orders of axes



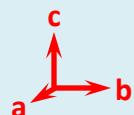
**2<sub>x</sub>**



**2<sub>y</sub>**



**2<sub>z</sub>**



2-fold axes in three  
perpendicular directions

$\alpha = \beta = \gamma = 90^\circ$

# Tetragonal

$P\ 4\ 2_12$   
 $P\ 4\ 2\ 2$   
 $I\ 4\ 2\ 2$

$P\ 4_12_12$   
 $P\ 4_12\ 2$   
 $I\ 4_12\ 2$

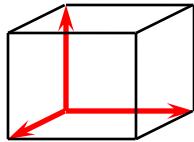
$P\ 4_22_12$   
 $P\ 4_22\ 2$

$P\ 4_32_12$   
 $P\ 4_32\ 2$

$P4$   
 $P4_1$   
 $P4_2$   
 $P4_3$

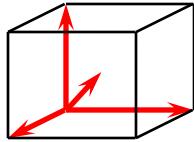
$I4$   
 $I4_1$

Lattice type



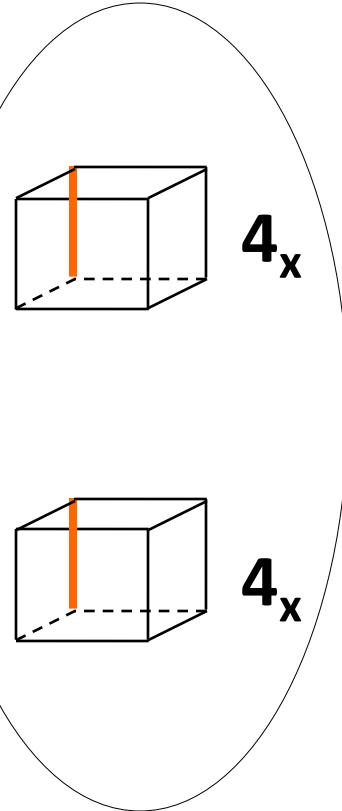
**P**

or

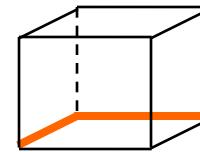


**I**

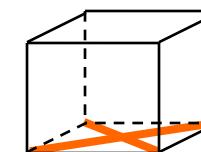
or



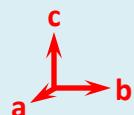
Directions and orders of axes



**2<sub>y</sub>**



**2**



4-fold axes along **c**

$\alpha = \beta = \gamma = 90^\circ$   
 $a = b$

# Cubic

$P\ 4\ 3\ 2$   
 $I\ 4\ 3\ 2$   
 $F\ 4\ 3\ 2$

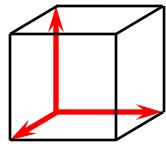
$P\ 4_13\ 2$   
 $I\ 4_13\ 2$   
 $F\ 4_13\ 2$

$P\ 4_23\ 2$   
 $P\ 4_33\ 2$

$P\ 2\ 3$   
 $I\ 2\ 3$   
 $F\ 2\ 3$

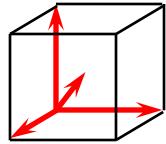
$P\ 2_13$   
 $I\ 2_13$

Lattice type



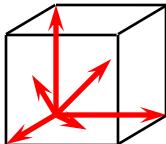
**P**

or



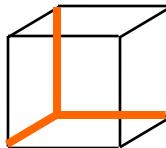
**I**

or



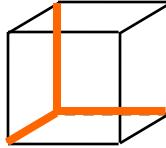
**F**

Directions and orders of axes

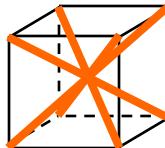


**4<sub>x</sub>**

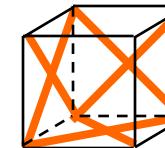
or



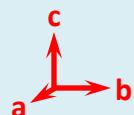
**2<sub>x</sub>**



**3**



**2**



3-fold axes parallel to  
the body diagonals

$\alpha = \beta = \gamma = 90^\circ$   
 $a = b = c$

# Trigonal - P

$P\ 3\ 2\ 1$

$P\ 3\ 1\ 2$

$P\ 3_1\ 2\ 1$

$P\ 3_1\ 1\ 2$

$P\ 3_2\ 2\ 1$

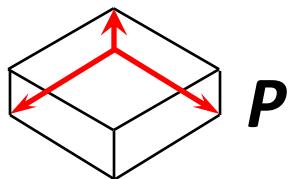
$P\ 3_2\ 1\ 2$

$P3$

$P3_1$

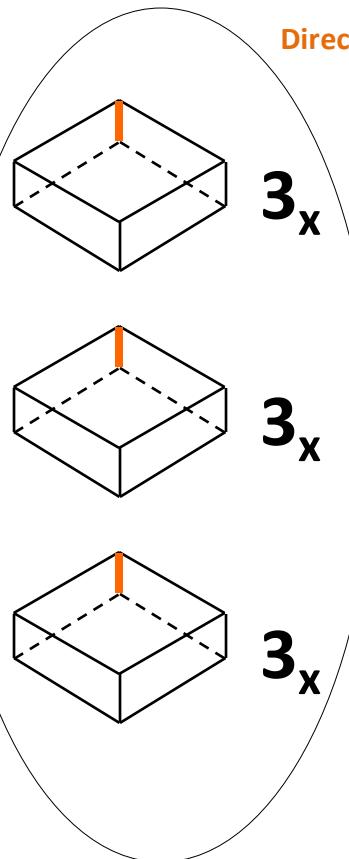
$P3_2$

Lattice type

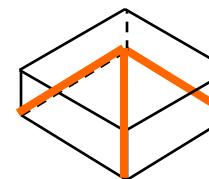


**P**

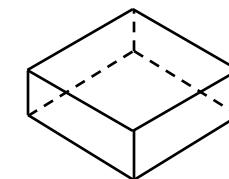
or



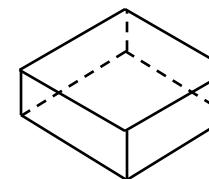
Directions and orders of axes



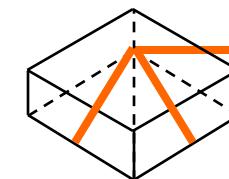
**2**



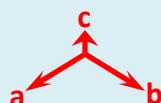
**1**



**1**



**2**



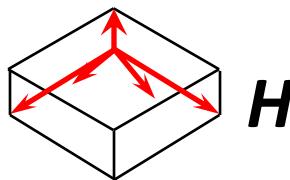
3-fold axes along **c**

# Trigonal - H

H 3 2

H3

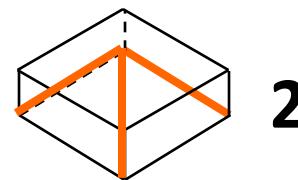
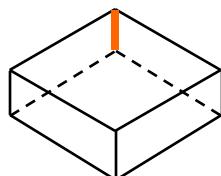
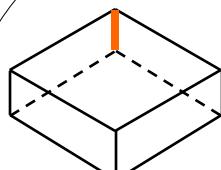
Lattice type



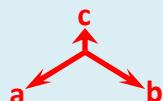
or

or

Directions and orders of axes



Rhombohedral setting **R** is an alternative to setting **H**



3-fold axes along **c**

# Hexagonal

P 6 2 2

P 6<sub>1</sub> 2 2

P 6<sub>2</sub> 2 2

P 6<sub>3</sub> 2 2

P6

P6<sub>1</sub>

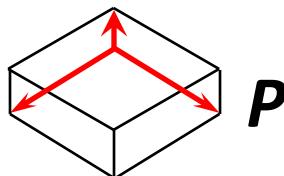
P6<sub>2</sub>

P6<sub>3</sub>

P6<sub>5</sub>

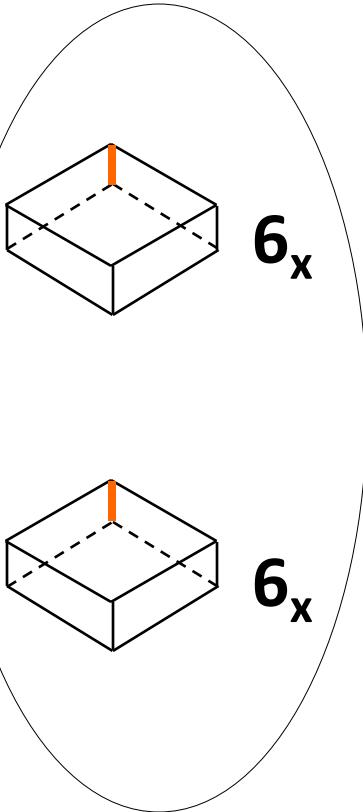
P6<sub>4</sub>

Lattice type

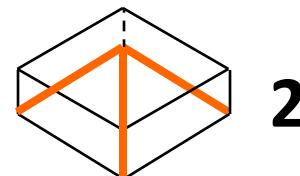


P

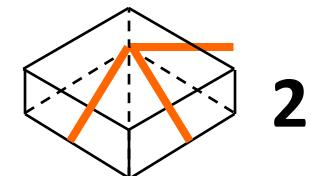
or



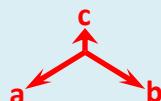
Directions and orders of axes



2



2



6-fold axes along c

$\alpha = \beta = 90^\circ$        $a = b$   
 $\gamma = 120^\circ$

# Ones

$\cancel{P \ 1 \ 1 \ 1}$

$P \ 1$

$P \ 1 \ 2 \ 1$

$P \ 2$

$\cancel{P \ 4 \ 1 \ 1}$

$P \ 4$

$\cancel{P \ 3 \ 1 \ 1}$

$P \ 3$

$P \ 3 \ 2 \ 1$

$\cancel{P \ 3 \ 2}$

$P \ 3 \ 1 \ 2$

$\cancel{P \ 3 \ 2}$

# Subscripts

P43212

P 43 21 2

P 4(3) 2(1) 2

$P4_32_12$

# Symmetry based setting vs. lattice based setting: C2

Symmetry based setting:	$\alpha = \gamma = 90^\circ$	C 1 2 1
Lattice based setting:	$\alpha = \gamma = 90^\circ$ $\beta < 120^\circ$	C 1 2 1 or I 1 2 1 - the same space group - different crystal setting

# Symmetry based setting vs. lattice based setting: primitive orthorhombic

Symmetry based setting:       $\alpha = \beta = \gamma = 90^\circ$        $P\bar{2}\bar{2}\bar{2}_1$        $P2_12_12$

Lattice based setting:       $\alpha = \beta = \gamma = 90^\circ$        $P\bar{2}\bar{2}\bar{2}_1$        $P2_12_12$   
                                 $a < b < c$        $P\bar{2}\bar{2}_1$        $P2_1\bar{2}2_1$   
                                 $P2_12\bar{2}$        $P2\bar{2}_12_1$

each of the two columns:  
- the same space group  
- different crystal setting

# Space group representation in ITC-A

Crystal Class  
(point group)

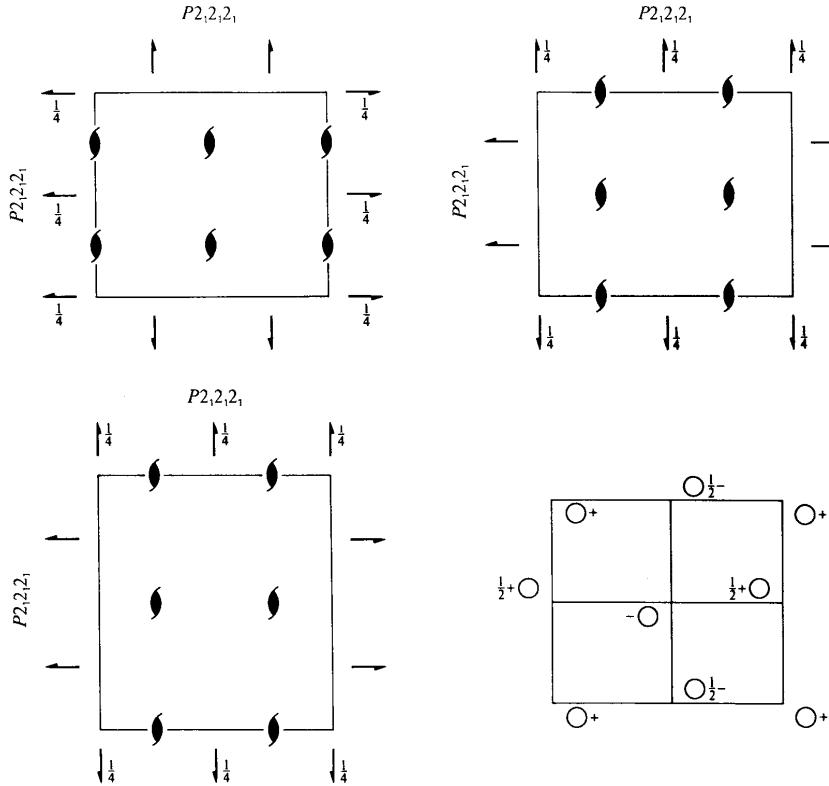
**222**

$P2_1 2_1 2_1$        $D_2^4$        $P2_1 2_1 2_1$

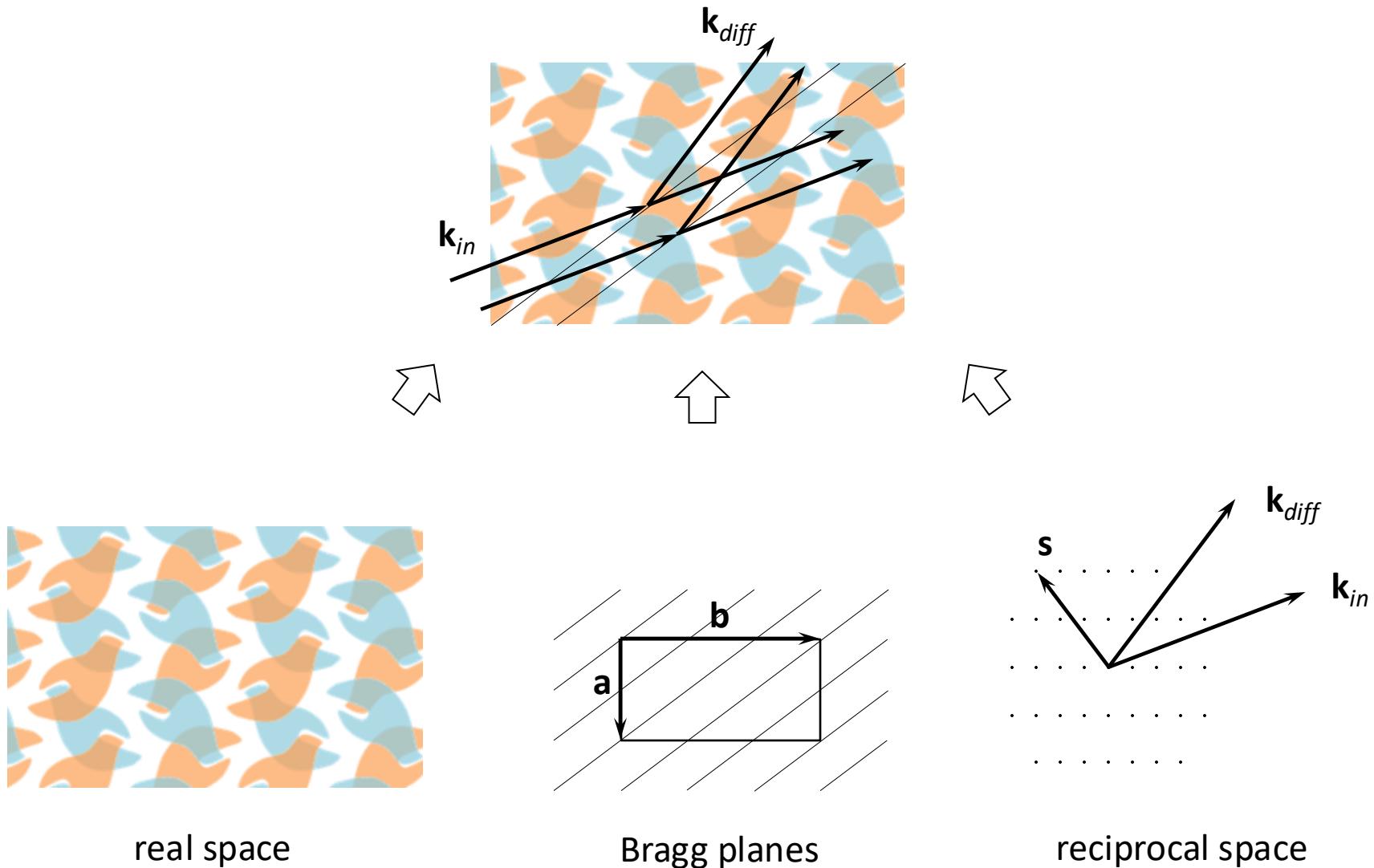
No. 19

Orthorhombic

Patterson symmetry  $Pmm$



# Conventional diffraction scheme



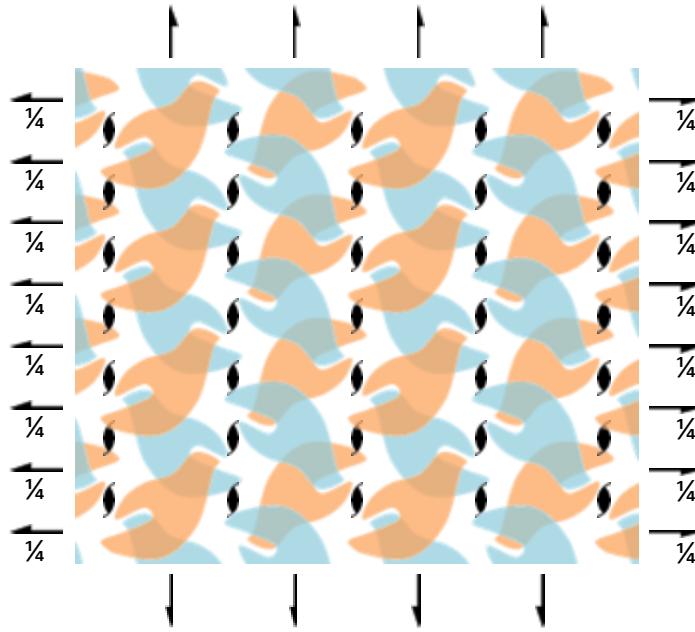
# Symmetry of intensities

The concept of reciprocal lattice is based on **angular** relations between the incident beam and the Bragg planes. Therefore:

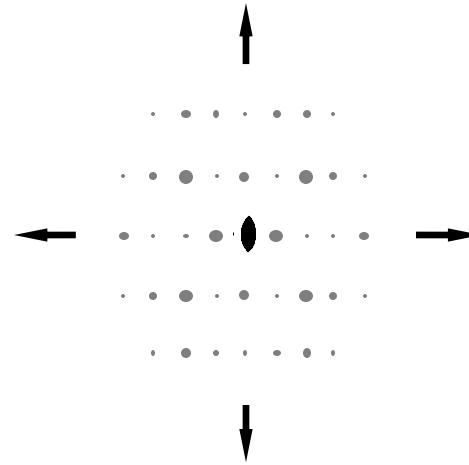
- Reciprocal lattice rotates together with crystal
- However, reciprocal lattice is not translated together with crystal

# Symmetry of intensities

real space



reciprocal space



All axes of the same order and in the same direction are "merged" together

to give an element of a point group.

# Symmetry of intensities

Real space

Crystal structure

Space group operation

The same crystal structure

Reciprocal space

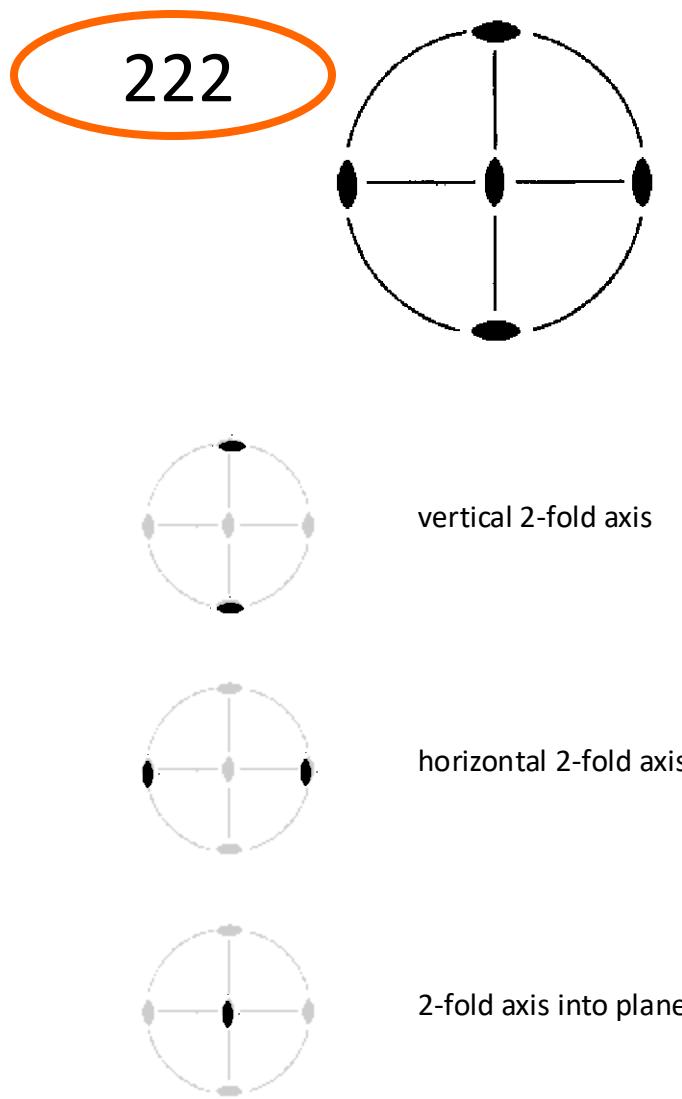
Intensities at Bragg points

Strip any translation component from the space group operation and merge identical operations:

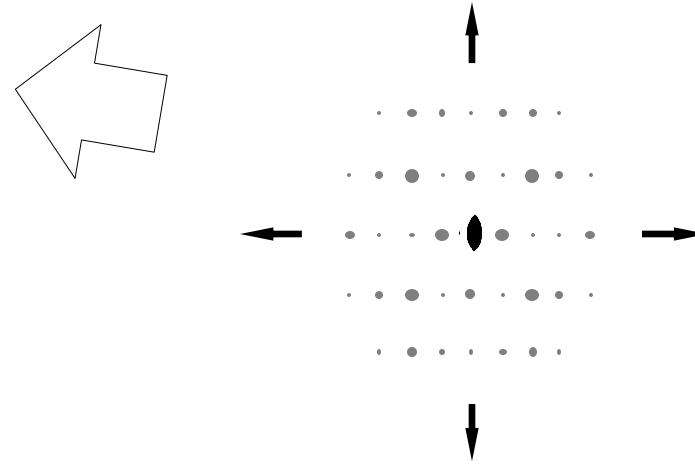
Point group operation

The same set of intensities (within  $\sigma$ )

# Point group scheme



reciprocal space



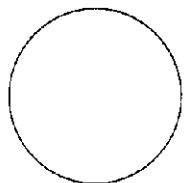
vertical 2-fold axis

horizontal 2-fold axis

2-fold axis into plane

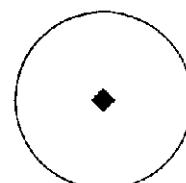
# The point groups that can exist in protein crystals

1

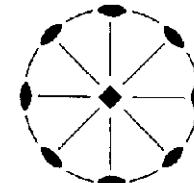


If it helps view as sphere

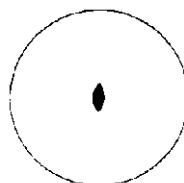
4



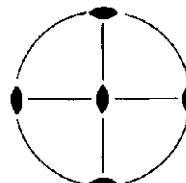
422



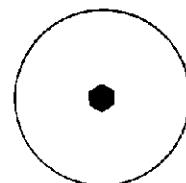
2



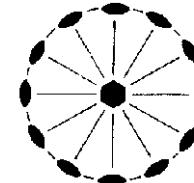
222



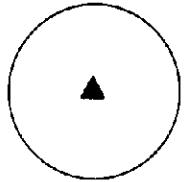
6



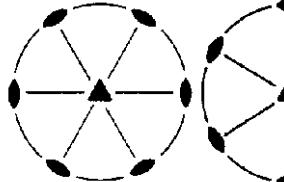
622



3



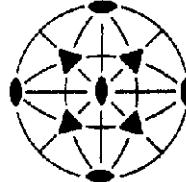
321



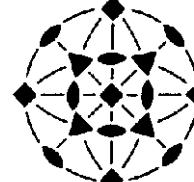
312



23



432



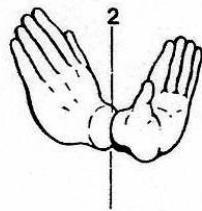
# The point groups that can exist in protein crystals

1

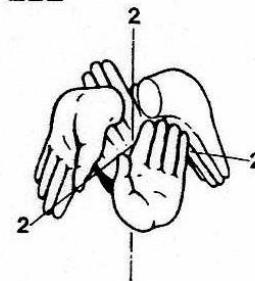


maybe an easier  
representation

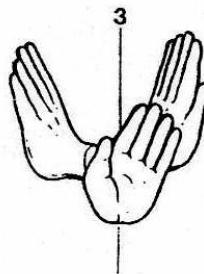
2



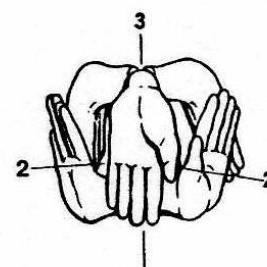
222



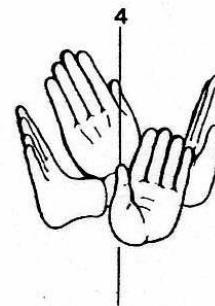
3



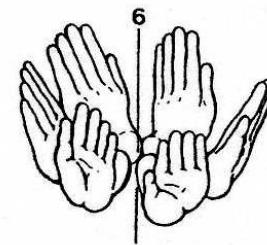
32



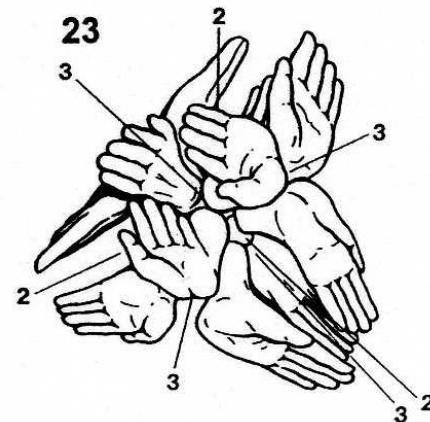
4



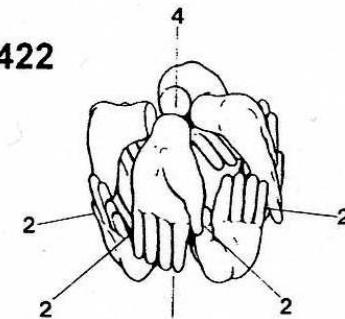
6



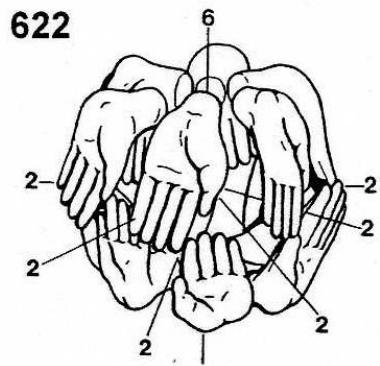
23



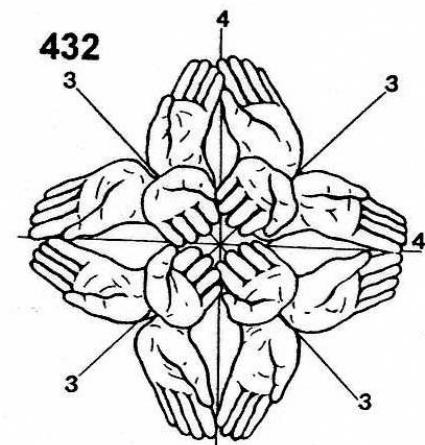
422



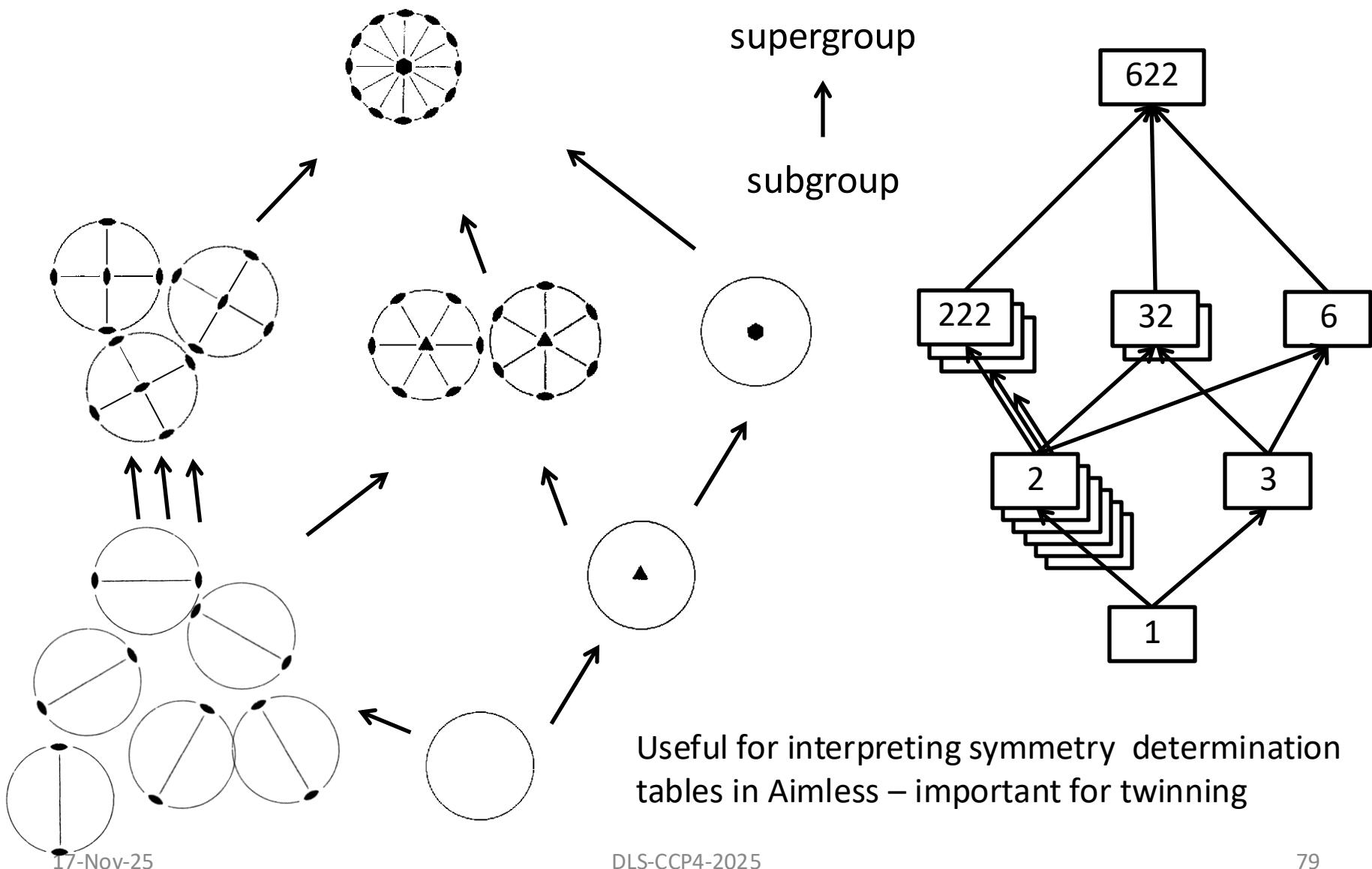
622



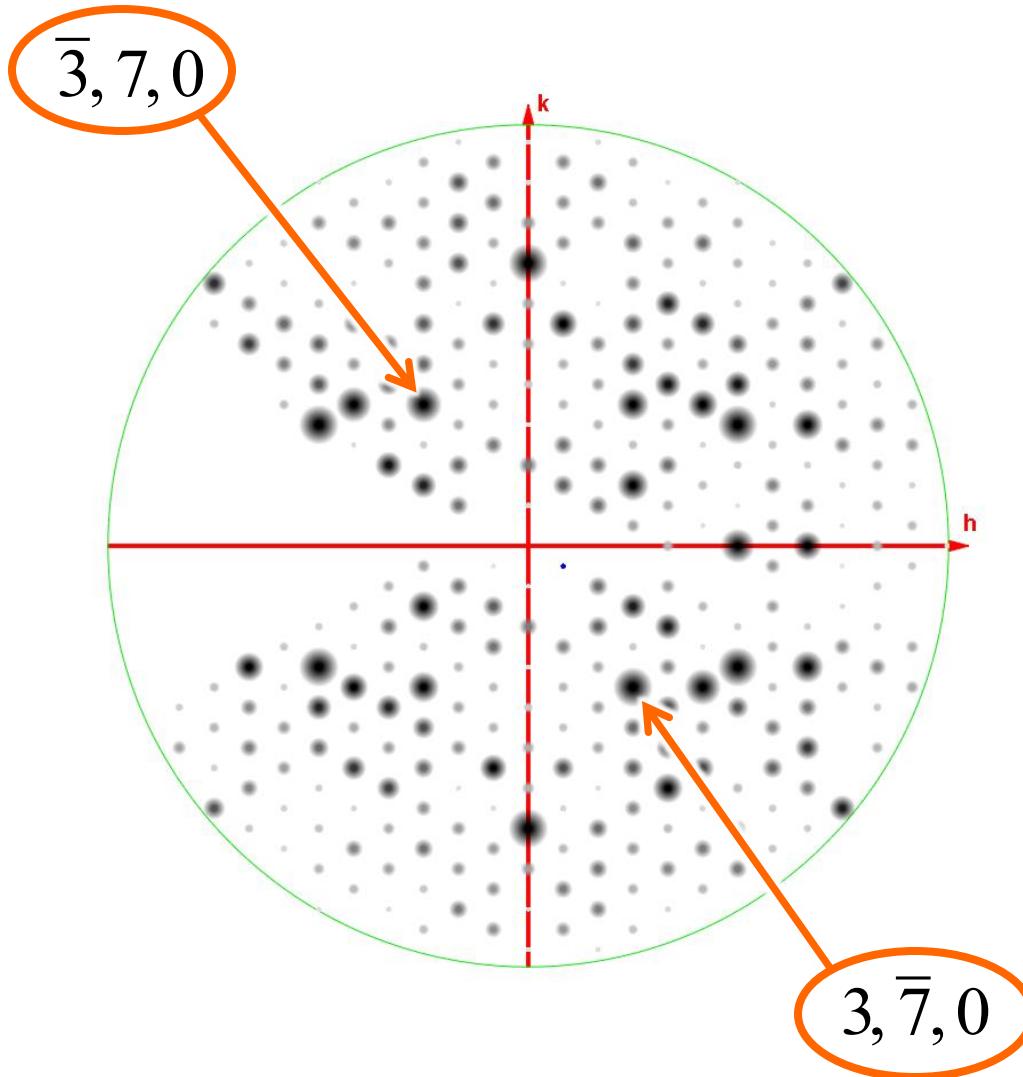
432



# Subgroups of the point group 622



# Friedel's law



(no anomalous signal)

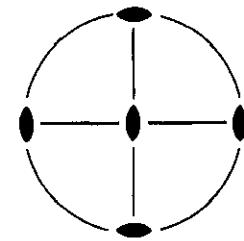
$$I(\bar{3}, 7, 0) = I(3, \bar{7}, 0)$$

# Point group and Laue group

+ inversion =

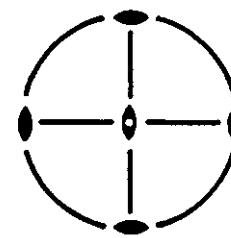
Crystal point group

222



Laue point group

m m m

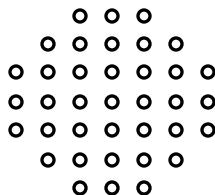


# The eleven Laue point groups or crystal classes

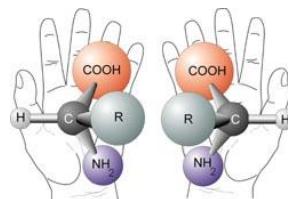
Crystal system	Laue point group experiment (no anomalous data)	Non-centrosymmetric point groups belonging to the Laue point group		
Cubic	$m3m$ $m3$	432	$\bar{4}3m$	
Tetragonal	$4/mmm$ $4/m$	23		
		422	$4mm$	$\bar{4}2m$
		4	$\bar{4}$	
Orthorhombic	$mmm$	222	$mm2$	
Trigonal	$3m$ $3$	32	$3m$	
		3		
Hexagonal	$6/mmm$ $6/m$	622	$6mm$	$\bar{6}m2$
		6	$\bar{6}$	
Monoclinic	$2/m$	2	$m$	
Triclinic	$\bar{1}$	1		

# Space group assignment (e.g. Pointless)

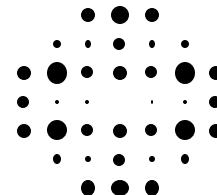
Reciprocal space lattice  
(positions of reflections)



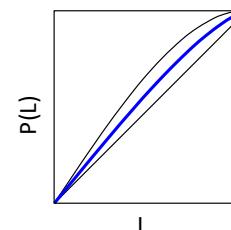
Mirror symmetry is not  
allowed in biological  
macromolecules



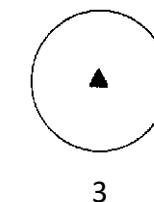
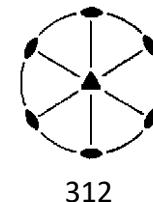
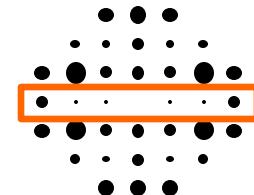
Intensities of  
reflections



Twinning  
test



Intensities of axial  
reflections



$P3_1$  or  $P3_2$

Lattice  
point group

Highest possible  
crystal point group

Approximate  
point group of  
diffraction

Possible  
crystal point  
group

Possible  
crystal space  
group(s)

User: decision making, structure solution, final space group assignment

# End

(Short) Hermann-Mauguin symbol

$P2_1 2_1 2_1$

No. 19

(Extended) Hermann-Mauguin symbol

$D_2^4$

$P2_1 2_1 2_1$

Crystal Class (point group)

222

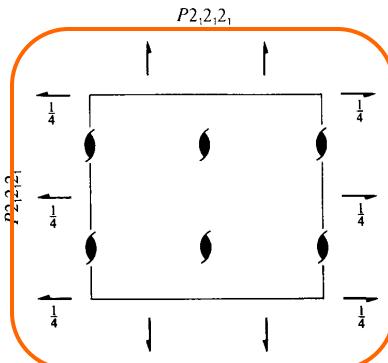
Crystal system

Orthorhombic

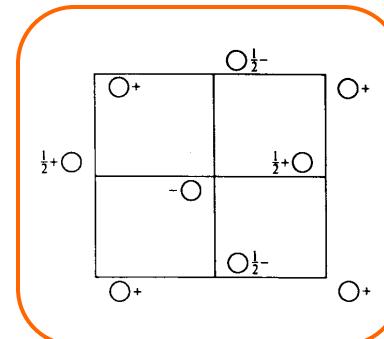
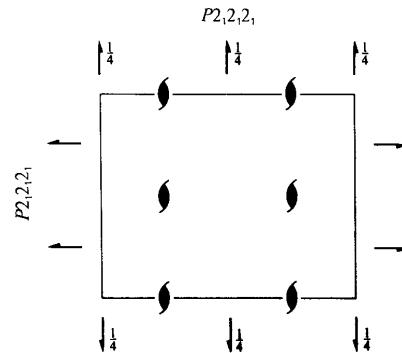
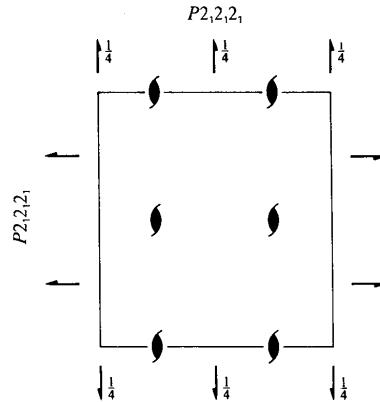
Patterson symmetry  $Pmmm$

Patterson symmetry

Location of symmetry elements



Two other projections are also shown for this space group



Set of equivalent points in general position.

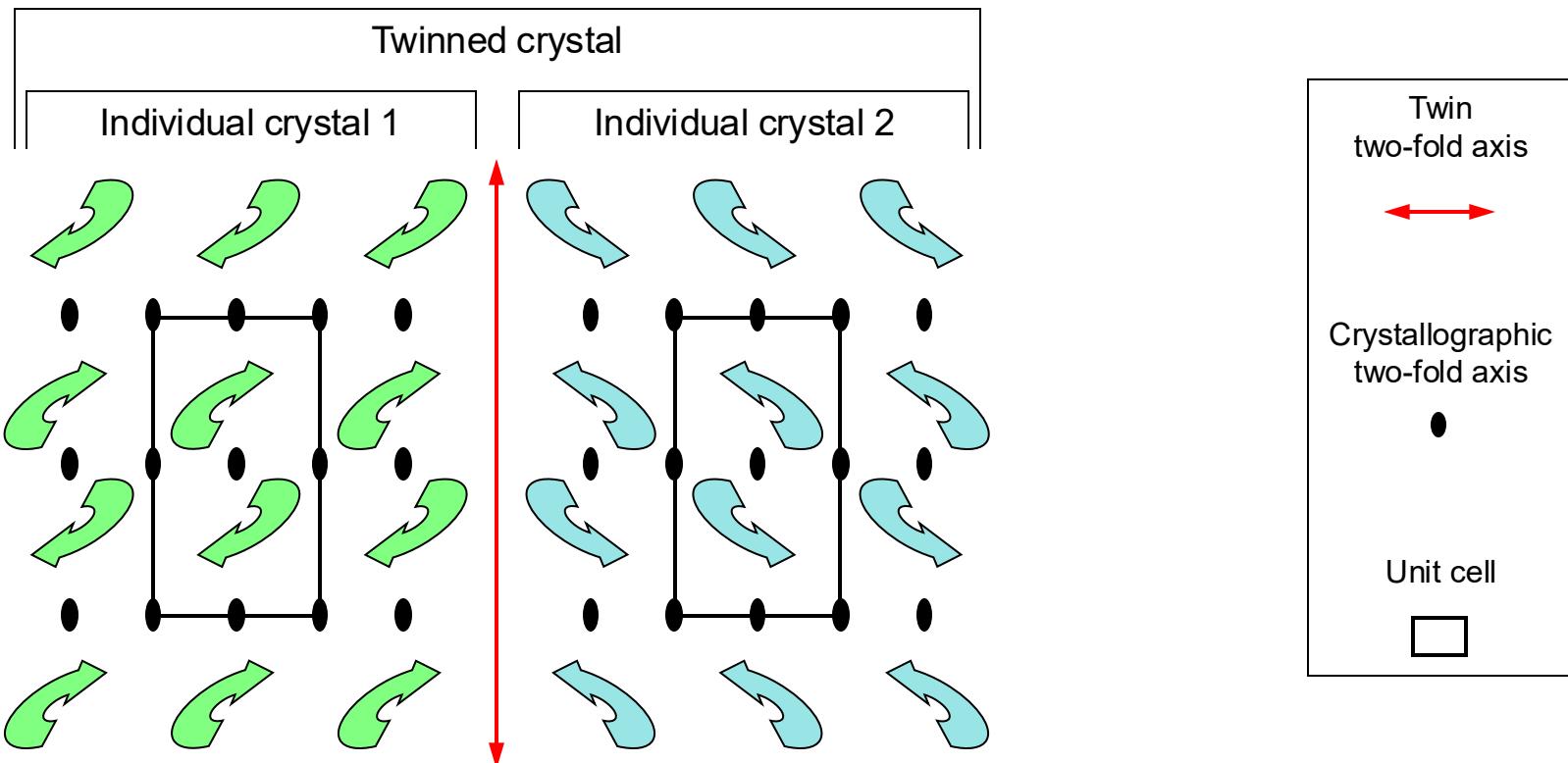
We were looking at "molecular wallpaper" instead



# Twinning by (pseudo)merohedry

lattice in orientation 1	=	lattice in orientation 2		
crystal in orientation 1	≠	crystal in orientation 2		
crystal in orientation 1	+	crystal in orientation 2	=	Twinned crystal

Example: P2,  $\beta = 90^\circ$

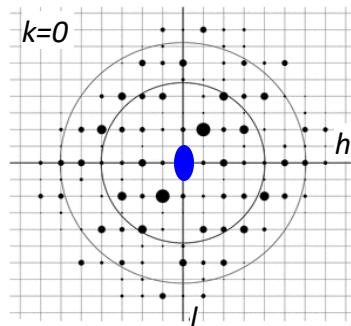


# Twinning by (pseudo)merohedry

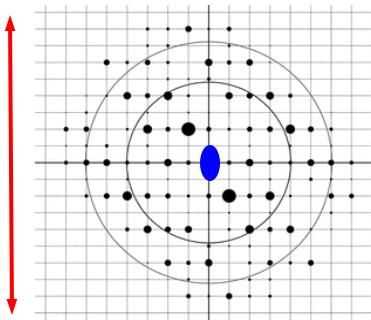
lattice in orientation 1	=	lattice in orientation 2		
crystal in orientation 1	≠	crystal in orientation 2		
crystal in orientation 1	+	crystal in orientation 2	=	Twinned crystal

P121,  $\beta = 90^\circ$

Intensities from  
individual crystal 1

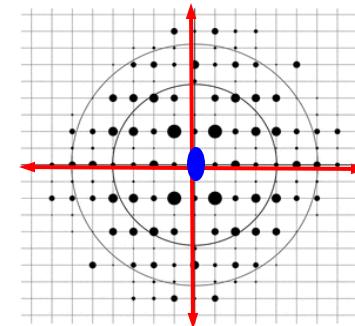


Intensities from  
individual crystal 2



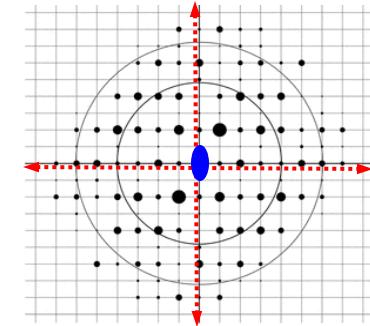
Perfect twin

(individual crystals  
of **equal** sizes)



Partial twin

(individual crystals  
of **different** sizes)



Twin two-fold axis



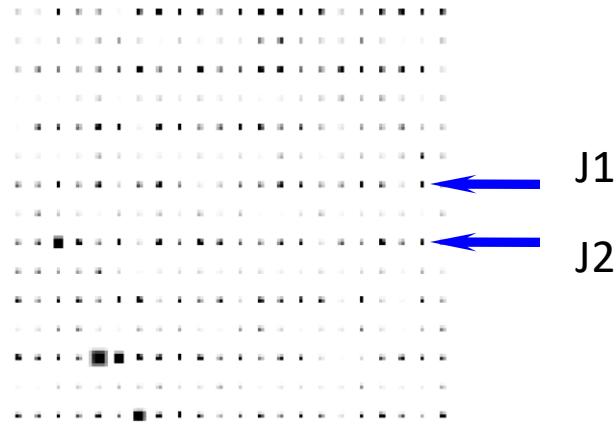
Crystallographic two-fold axis

less weak reflections

Allows for twinning tests based on statistics of intensities

# L-test

$$L = | J_1 - J_2 | / ( J_1 + J_2 )$$

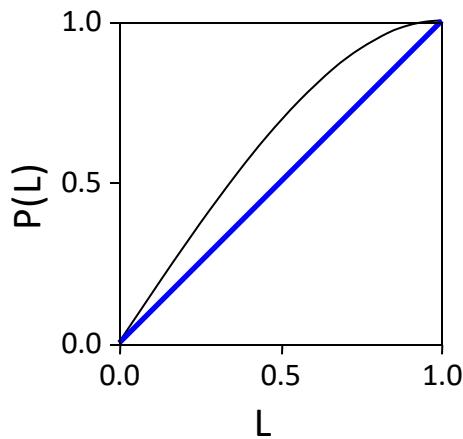


L-test is designed to be suitable for most of cases

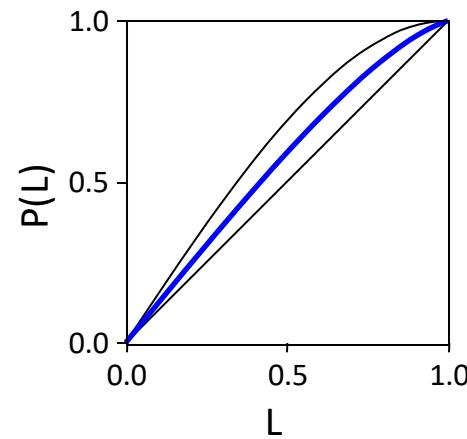
- with and without pseudo-translation
- isotropic and anisotropic data

# Theoretical distribution of L

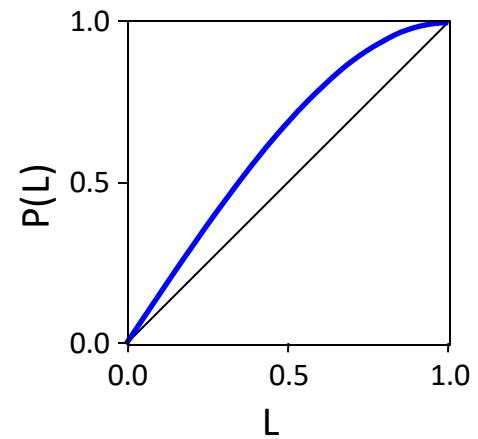
Single crystal



Partial twin



Perfect twin



# Purpose of twinning tests

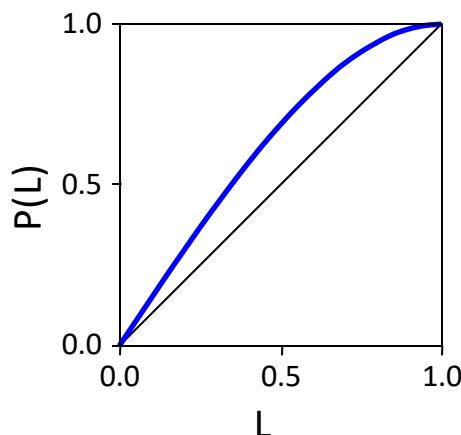
Twinning tests only indicate possible issues with crystal symmetry

All the rest belongs to refinement

- twinning fraction estimation
- evaluation of R-factors
- “detwinning” for map calculation
- final decision on crystal symmetry

# Example

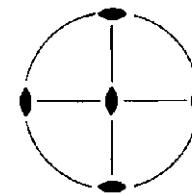
L-test



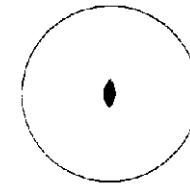
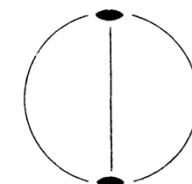
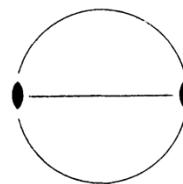
Perfect twin



Symmetry of diffraction\* 222, lattice type P



Point group symmetry of the crystal:  
a subgroup of 222 (one of the point groups 2):



Space groups to try by MR\*\*:

P121

P12<sub>1</sub>1

P211

P2<sub>1</sub>11

P112

P112<sub>1</sub>

# A wake-up slide

Attached is a zipped directory with the sequence file (...) of the crystallised protein (pyruvate kinase from *Methanosa*cina *thermophila*), merged data used in refinement (...), and results of my last refinement (...).