

1 **The color rendering and near infrared reflection properties of**
2 **coated iron oxide green pigments**

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5

6 **Abstract**

7 In this study, a novel iron oxide green pigment (FePc), composed of phthalocyanine
8 blue (CuPc) and iron oxide yellow pigments (FeOOH), was prepared by intermittent
9 stirring. The FePc was coated with the hydrolysate of sodium metasilicate and silane
10 coupling agent KH602. The structure and properties of prepared pigments were
11 characterized by TEM, SEM, XRD, FT-IR, XPS, and TG-DTA techniques. The
12 aspect ratio of FePc decreased compared to FeOOH. XPS analysis showed that CuPc
13 and FeOOH were bound by hydrogen bonds. The thermal decomposition temperature
14 of the coated pigments increased from 266 °C to 314 °C, and the color difference ΔE
15 of pigment was 1.623 after baking at 240 °C for 30 min, indicating the prepared
16 pigments had excellent heat resistance. The surface modification of FePc improved
17 their hydrophobicity and dispersibility in organic media. The pigments with
18 CuPc/FeOOH ratio of 0.2 exhibited well color properties ($L^*=55.44$, $a^* = -9.9$, $b^* =$
19 6.63), and green tone. The coating and surface modification had little impact on a^*
20 value. Particularly, all the newly prepared green pigments displayed high NIR
21 reflectance. The prepared pigments have high heat resistance and dispersion
22 properties, and are expected to be used in fields such as plastics and high-temperature
23 coatings.

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24 **Keywords:** iron oxide yellow pigment, phthalocyanine, green pigment, coated, heat
25 resistance

26 **1. Introduction**

27 The existence of natural dyes and pigments makes the world rich and colorful.
28 However, natural dyes or pigments extracted from plants and animals exhibit a lack of
29 stable color tone, poor chemical durability and thermal stability when subjected to
30 heating, light, and chemical substances [1]. As a result, people have also attempted to
31 synthesize pigments with different colors through various methods. Various shades of
32 green pigment are reported such as uvarovite ($\text{Ca}_3\text{Cr}_2(\text{SiO}_4)_3$)[2], Cr-doped BiPO_4 [3],
33 $\text{Cr}_2\text{O}_3/\text{SiO}_2$ composite pigments [4]. These green pigments contain chromium ions.
34 More importantly, chromium ions exhibit five valence states from +2 to +6, which
35 transition to divalent or trivalent cations as the environment changes, ultimately
36 leading to uncertainty in the color of the pigment [5]. In addition, among various
37 oxides, chromium has multiple oxidation states, and Cr single bonds and O valence
38 bonds can change under changes in temperature and/or oxygen partial pressure,
39 thereby affecting the color tone of Cr based pigments. In actual production, the color
40 tone of Cr based pigments is also difficult to control [6]. Other green pigments such as
41 Scheele's green (CuHAsO_3) and Emerald green ($\text{Cu}(\text{CH}_3\text{CO}_2)\text{AsO}_3$), despite their
42 high light-fastness, excellent price-performance ratio, brilliant hue, and high thermal
43 stabilities [7], contain arsenic, making them undesirable. More importantly, inorganic
44 green pigments are difficult to use in the high-temperature paint coating field. In
45 addition, organic green pigments mainly consist of phthalocyanine green, which has
46 well light and weather resistance, strong coloring power, bright color, small color
47 difference, good color spreading and fluidity [8, 9]. However, the price of
48 phthalocyanine green pigment is relatively high. Therefore, there is an urgent need to

49 develop environmentally friendly and cost-effective green pigments.

50 Preparing green pigments through color blending is also a new approach. Through
51 the subtractive method in the principle of color mixing, it is found that green pigments
52 can be prepared by blending yellow and blue pigments. Commonly used yellow
53 pigments include doped ZrO₂, (Bi,Ca,Zn)VO₄, BiVO₄-GdPO₄, Ce substituted SrMoO₄
54 [10-13]. Goethite (α -FeOOH) is a common and stable iron oxide present in natural
55 environment [14]. So far, due to the advantages of chemical stability, non-toxic and
56 low cost, α -FeOOH has been extensively studied in water purification, organic
57 pollutant degradation, coatings, pigments and other fields [15-18]. Regarding blue
58 pigments, there are Han blue (BaCuSi₄O₁₀) and Maya blue used a long time ago [19],
59 as well as Cerulean blue (CoSnO₃) and Prussian blue ($\text{Fe}_4[\text{Fe}(\text{CN})_6]_3$) used in
60 currently[20-22]. Another type of blue pigment is organic blue pigment, such as
61 phthalocyanine blue. Phthalocyanine blue is an important organic pigment with bright
62 colors and strong coloring power [23, 24]. The preparation of green pigments using
63 inorganic iron oxide yellow pigment and organic blue pigment phthalocyanine blue as
64 raw materials is very challenging. It is necessary to consider the combination between
65 organic and inorganic pigments and the color rendering performance.

66 It's important to note that ordinary iron oxide yellow pigment α -FeOOH begin to
67 dehydrate at 177 °C and continue to dehydrate to become Fe₂O₃ between 270 °C and
68 320 °C, thus changing the color of pigment and affecting its application in high
69 temperature occasions [25, 26]. Therefore, it is necessary to improve the heat
70 resistance of the new green pigments synthesized from α -FeOOH, so that it can meet
71 the requirements of the coating of the coil material with the heat-resistant temperature
72 between 210 and 250 °C. On the other hand, it is also necessary to improve the
73 compatibility of synthesized green pigments with organic matrix and the wettability

74 of the matrix in order to broaden the application of synthesized green pigments in
75 high temperature coating industry such as plastic processing and coil coatings.
76 Recently, the focus of pigments formulators has shifted to develop materials with high
77 near-infrared reflectivity (NIR) while maintaining their appropriate spectroscopic
78 features that impart colour in the visible spectrum [27, 28]. The NIR reflective
79 pigments have significant advantages in thermal protection of containers, ceiling
80 coatings, and storage room surfaces to reduce the use of air conditioning.

81 In this work, inorganic FeOOH was used as the core, organic phthalocyanine blue
82 pigment was coated externally to prepare composite green pigment (FePc). The FePc
83 was modified to improve its heat resistance, and finally surface modification was
84 carried out with a coupling agent to improve the dispersion performance of pigments
85 in organic media. Besides, the prepared pigments were characterized by multifarious
86 tools to study crystal structure, morphological characteristics, color coordinate values
87 and optical performance. In addition, studied the heat resistance, coloration, and
88 dispersion properties of pigments to evaluate their application. More importantly, the
89 coloration mechanism and near-infrared reflectance performance based on spectral
90 analysis have been further investigated.

91

92 **2. Experimental Section**

93 *2.1 Materials*

94 Phthalocyanine blue pigments (CuPc, C.I.pigment bule 15: 3), were purchased
95 from Shandong Yousuo Chemical Technology Co., Ltd. Iron oxide yellow pigments
96 (FeOOH), were provided by Zhejiang Huayuan Pigments Co., Ltd. Sodium silicate,
97 dilute sulfuric acid, silane coupling agent KH602, etc. were analytical purity and were
98 purchased from East China Pharmaceutical Co., Ltd. All solutions were prepared with

99 distilled water.

100 *2.2 Preparation of coated iron oxide green pigments*

101 Weigh the pigments according to the mass ratio of FeOOH to CuPc was 5:1. The
102 pigment was put into a high-speed pulverizer for mixing. After mixing for 1min, stop
103 for 1 min, and then continue mixing. The final mixing time was 10 min. The iron
104 oxide green pigment was obtained and marked as FePc.

105 The preparation of coated FePc was follows: FePc (10g) was added in 350 mL
106 distilled water, then dispersing for 10 min with ultrasound to get suspension. Under
107 water bath at 70 °C, the aboved suspension was stirred at 300-400 rpm. Certain
108 quality of sodium silicate solid, which was calculated by the mass of silicon dioxide
109 and the mass was 8% of the mass of FePc, was put into another beaker, and 200 mL
110 of water was added to make it completely dissolved. Then the above solution was
111 placed in a constant pressure funnel and dropped into the FePc suspension at a
112 constant rate of 1 d/s, with a dripping time of about 60 min. After dropping, adjust the
113 pH value of the solution to 8 with sulfuric acid, and continue stirring for 0.5 h. During
114 this process, the pH value remained unchanged. Finally, the precipitate was filtered,
115 washed with distilled water for three times and then dried at 70°C for 15 h. After that,
116 the above product was identified as FePc@SiO₂.

117 Coated FePc@SiO₂ was modified with silane coupling agents KH602. 5g
118 FePc@SiO₂ was added in 250 mL distilled water, then dispersing for 10 min to get
119 suspension. In water bath 80 °C, the aboved suspension was stirred at 300-400 rpm.
120 The KH602 (8%, based on the quality of FePc@SiO₂) was dissolved in 50mL
121 distilled water, slowly dropped into FePc@SiO₂ suspension, and stirred for 90 min.
122 Finally, the precipitate was filtered, washed with distilled water for three times and
123 then dried at 70°C for 15 h. After that, the above product was marked as

124 FePc@SiO₂-M.

125 *2.3 Characterizations*

126 Scanning electron microscopy (SEM) analysis was performed on a Hitachi Model
127 S-3400N electron microscope operated at an accelerating voltage of 15 kV.
128 Transmission electron microscopy (TEM) images were taken on a FEI Tecnai G2 F20
129 electron microscope. X-ray diffractometer (Cu K α , $\lambda=0.1545$ nm, pipe voltage 36 kV,
130 pipe current 20 mA) (Beijing Purkinje General Instrument Co., Ltd., China). Infrared
131 spectra were recorded using the KBr disc method on a 5700 FT-IR spectrophotometer
132 (Nicolet, USA). X-ray photoelectron spectroscopy (XPS) spectra of the solids were
133 collected using a 250XI K-alpha spectrometer (Thermo Fisher Scientific), with a
134 monochromatic Al Ka radiation source and a hemi-spherical electron analyzer.
135 Survey scans were recorded with a 500 μ m spot size and a fixed energy pass of 100
136 eV, where energy step size was 1.000 eV. The thermal properties were measured by
137 simultaneous thermal analyzer SDT Q600 (TA instruments, USA), heating from the
138 room temperature up to 400 °C with a heating rate of 10 °C/min. The wettability of
139 sample is evaluated by SZ-CAM series contact angle tester (Xuanzhun instrument).
140 Water droplets were deposited directly on the surface of the material and the water
141 contact angles (WCA) were measured. Three measurements were performed per
142 sample and averaged. The volume of the water droplet was 5 μ L and the tip used was
143 a precision stainless steel tip.

144 *2.4 Color and UV-Vis-NIR measurements*

145 The visible light absorption spectrum and NIR region (700–2500 nm) was recorded
146 by the UV-Vis-NIR spectrophotometer (Lambda 1050+, Perkin-Elmer). The CIE
147 LAB color data (L^* , a^* , and b^*) were calculated from the visible light reflection
148 spectrum by color CIE software (Perkin-Elmer, CIE D65 photo source, and 10

149 observation angle; the calculated spectrum range was 400–700 nm)[30]. The NIR
150 reflectance (R) in the wavelength range of 700 nm to 2500 nm can be calculated by
151 the following formula:

$$152 R = \frac{\int_{700}^{2500} r(\lambda) i(\lambda) d\lambda}{\int_{700}^{2500} i(\lambda) d\lambda} \quad (1)$$

153 where $r(\lambda)$ is the spectral reflectance ($\text{W}\cdot\text{m}^{-2}$) obtained from the UV–Vis–NIR
154 spectrophotometer and $i(\lambda)$ is the solar spectral irradiance ($\text{W}\cdot\text{m}^{-2}\cdot\text{nm}^{-1}$) obtained
155 from ASTM standard G159-98 [29]. NIR reflectance value was used to quantify the
156 solar radiation, evaluate their reective performance.

157 Calculation of the color difference (ΔE) after baking was used to evaluate the color
158 stability of pigments. The sample (2.0 g) was put into crucible, and baked at 240 °C
159 for 30 min. Then taken out and cooled to room temperature to measure the value of
160 color difference. According to Hunter's color difference formula [31]:

$$161 \Delta E = (\Delta L^2 + \Delta a^2 + \Delta b^2)^{1/2} \quad (2)$$

162 Usually, ΔE is the value of color difference, ΔL is the lightness axis (positive is for
163 white and negative is for black). The parameters Δa (positive is for red and negative is
164 for green) and Δb (positive is for yellow and negative is for blue) indicate the hue or
165 color of the pigments.

166

167 **3. Results and discussion**

168 *3.1 Morphology analysis*

169 **Insert Fig. 1**

170 SEM images of FeOOH and the prepared FePc are displayed in Fig.1. The color of
171 FeOOH (Fig.1a) is obvious yellow, and it has a typical needle-like structure with a
172 length of about 2 μm , and a small amount of agglomeration between particles. When

173 CuPc and FeOOH are blended to prepare FePc, the FePc pigment shows obvious
174 green. The FePc are mostly rod-shaped, with a reduced aspect ratio and a reduced
175 agglomeration phenomenon. Comparing the morphology, the particle size and length
176 of FePc decreased, which may be caused by high speed stirring and shearing during
177 the preparation process.

178 Fig. 1c shows the multi-elemental EDS mapping images of C, N, O, Fe, and Cu in
179 FePc. The O and Fe elements come from FeOOH, the N element mainly comes from
180 N in the molecular structure of CuPc. Apparently, the corresponding N and Cu maps
181 illustrate a homogeneous distribution of these elements in the field of the
182 cross-section view. At the same time, it is also certified that the uniform distribution
183 of CuPc in iron oxide green pigment.

184 To further confirm the formation of the FePc@SiO₂ interface before and after
185 modification, the TEM analysis were performed. Fig. 2(a) is the TEM diagram of
186 FePc coated with silica. The surface of FePc is uniformly coated with a thin coating
187 layer, and the thickness between 3.2 and 4.3 nm. This layer may be SiO₂, indicating
188 that SiO₂ has been uniformly coated on FePc surface.

189 The lattice fringes of FeOOH are clearly visible, and the spacing of (110) planes is
190 calculated to be 0.42 nm (Fig. 2b). From Fig.2c and Fig.2d, the KH602 is modified
191 successfully on FePc@SiO₂ surface, the coating layer is uniform, and the thickness of
192 coating layer is about 6 nm. In conclusion, the coating modification of FePc is
193 successfully realized using SiO₂ and KH602.

194 **Insert Fig. 2**

195 *3.2 XRD Characterization*

196 **Insert Fig. 3**

197 The XRD patterns of FePc, FePc@SiO₂ and FePc@SiO₂-M are shown in Fig.3.

198 The diffraction peaks at about $2\theta=17.8^\circ$, 21.2° , 34.7° , 35.5° , 36.6° , 53.1° and 59.0° are
199 correspond to crystal planes (020), (110), (021), (101), (111), (221) and (151),
200 respectively, which are consistent with the characteristic diffraction peaks and crystal
201 planes of FeOOH[32]. Using the Bragg's formula ($2dsin\theta = n\lambda$) to calculate the
202 interlayer space d_{110} value of FeOOH, the d_{110} value is 0.4199nm. The diffraction
203 peaks of FePc and FeOOH are basically the same, and there are strong diffraction
204 peaks near $22^\circ(2\theta)$, $33^\circ(2\theta)$ and $37^\circ(2\theta)$, with sharp peak patterns. This shows that
205 FePc has well crystal form and high crystallinity, which means that the original
206 crystal structure of FeOOH is not changed after compounding with CuPc.

207 Compared with the diffraction peak of FePc, there is no obvious characteristic peak
208 of SiO_2 in FePc@ SiO_2 , which means that coating layer SiO_2 exists in amorphous state
209 [33]. Furthermore, the positions of diffraction peaks of FePc@ SiO_2 -M are basically
210 the same as that of FePc@ SiO_2 . The crystal structure of FePc did not change during
211 the coating and surface modification processes.

212 *3.3 FT-IR Characterization*

213 **Insert Fig. 4**

214 The FTIR spectra of FeOOH, FePc@ SiO_2 and FePc@ SiO_2 -M are shown in Fig. 4.
215 The absorption peaks (Fig.4a) of FeOOH at 3420 cm^{-1} are owing to surface adsorbed
216 water, and the strong absorption peak at 3125 cm^{-1} is attributed to the stretching
217 vibration of –OH in FeOOH. Four absorption peaks near 899, 796, 615 and 475 cm^{-1}
218 are characteristic absorption peaks of FeOOH [34]. The absorption peaks of CuPc at
219 1723 cm^{-1} and 1606 cm^{-1} are belong to the vibrational absorption peaks of C=C and
220 C=N in the phthalocyanine ring [35]. By analyzing the infrared spectrum of FePc, it is
221 found that the above two absorption peaks shift to 1799 and 1663 cm^{-1} . This result
222 may be due to the change of the electron cloud in the phthalocyanine ring, which is

223 caused by the physical and chemical interactions of the CuPc on FeOOH. Moreover,
224 the absorption peak at about 3420 cm⁻¹ disappeared, but the absorption peak at 3125
225 cm⁻¹ does not change, indicating that there are still hydroxyl groups on FePc surface.

226 Compared with FePc@SiO₂ (Fig.4b.), the absorption peak position of
227 FePc@SiO₂-M at 3100 cm⁻¹ changed slightly. This may be due to the change of the
228 hydroxyl group on the surface of FePc@SiO₂, resulting in the change of the stretching
229 vibration absorption peak of -OH group. However, the change of FT-IR spectrum
230 before and after modification with KH602 is not obvious, which may be caused by
231 less coating amount.

232 *3.4 XPS analyses*

233 The XPS spectra with full-scan of prepared pigments are shown in Fig.5a. The
234 binding energies at about 724.7 eV, 711.5 eV, 529.7 eV, 398.5 eV, 285.7 eV, 154.15
235 eV, 103.5 eV, 91.1 eV and 52.7 eV are consistent with the electronic states of Fe2p,
236 O1s, N1s, C1s, Si2s, Si2p, Fe3s and Fe3p, respectively [36]. The XPS spectrum of
237 FeOOH shows three elements: O, C, and Fe. According to the chemical formula, only
238 O and Fe belong to FeOOH, while C comes from excess hydrocarbons in the
239 environment. The Auger resonance peak of O is in Fig.5a, but the area of the peak is
240 not large, indicating that a small amount of O₂ is attached to FeOOH surface. The
241 characteristic peaks of N1s, Cu2p1 and Cu2p3 in CuPc molecule appear in the XPS
242 spectrum of FePc, and the intensity of the characteristic peaks of C1s increases. This
243 is mainly due to the ring structure of C, N and the central metal Cu in CuPc. When
244 FePc is coated with sodium silicate (FePc@SiO₂), it can be clearly seen that the peaks
245 of Si2s and Si2p are between 100-200 eV. When coated with KH602, the percentage
246 of C and Si in FePc@SiO₂-M decreases clearly, but the percentage of O significantly
247 increases. This is probably related with the elements of KH602 and atomic

248 composition of the interface layer of FePc@SiO₂-M.

249 The XPS spectrum of Cu2p (Fig.5b) shows that the binding energys of Cu2p_{3/2} in
250 FePc, FePc@SiO₂ and FePc@SiO₂-M are 935.05 eV, 935.11 eV and 935.21 eV,
251 respectively. It is found that the binding energy of Cu2p_{3/2} increases slightly,
252 indicating that the density of electron cloud around Cu atom increases, which is
253 caused by the effect of SiO₂ and KH602. Furthermore, compared with the C1s
254 binding energy of 285.7 eV, binding energy of C1s in FePc@SiO₂ (Fig.5c) is almost
255 unchanged. It shows that the binding mode between CuPc and FeOOH remains
256 unchanged after SiO₂ coated. After modification with HK602, the binding energy of
257 C1s is changed from 285.7 eV to 285.4 eV, which is mainly attributed to the C-O in
258 HK602. This indirectly confirms that the success of surface modification of
259 FePc@SiO₂ with HK602.

260 The tested samples have N1s main nitrogen peak at about 398.6 eV (Fig.5c-e),
261 which belongs to C-N bond. In addition, there is a weak $\pi-\pi^*$ satellite feature of the
262 shift several electronvolt of nitrogen-containing aromatic substances from the main
263 nitrogen peak near 400.3 eV [37]. The N1s binding energy in FePc@SiO₂ is 0.03 eV,
264 which is higher than that of FePc, implying the intermolecular force increases with the
265 increase of the electron cloud near N1s. The N element mainly comes from CuPc,
266 indicating that there is intermolecular force between the coating layer and CuPc after
267 coating. In additon, the binding energy of N1s about FePc@SiO₂-M continues to
268 increase (398.72 eV), which is due to the C-N bond in the molecular structure of
269 KH602. The above experimental results show that both coating and surface
270 modification can make the electron cloud near the N1s of FePc@SiO₂-M increases,
271 and the intermolecular force further increases.

272 **Insert Fig. 5**

273 **Insert Fig. 6**

274 As shown in Fig.6(a), about FeOOH, there are 3 peaks in the O1s fitting diagram,
275 and their binding energies are 529.65 eV, 531.05 eV and 533.21 eV, respectively.
276 Furthermore, the electron binding energy at 529.65 eV and 531.05 eV are ascribed to
277 the O in OH of FeOOH and O in FeOOH, respectively, the results fit the feature
278 structure of FeOOH [38]. The binding energy at 533.21 eV is from the C-O bond of
279 air. Moreover, as shown in Fig.6(b), the O1s binding energy in FePc increases,
280 compared with that in FeOOH, which shows the ability of oxygen to lose electrons in
281 FePc is slightly reduced. According to the FT-IR analysis, the absorption peak of-OH
282 in FePc does not shift. Therefore, the shift of binding energy of O1s may be because
283 of the formation of hydrogen bond between O in FeOOH and H in the -C-H group of
284 CuPc ring, resulting in the change of electron cloud near the C=C and C=N groups in
285 phthalocyanine ring.

286 In Fig.6(c), there are also 3 peaks in the O1s fitting diagram in FePc@SiO₂, and
287 their binding energies are 529.9 eV, 531.21 eV and 532.58 eV, respectively. Also,
288 these 3 binding energies are respectively relative to the O in FeOOH, O in FeOOH,
289 and O in C-O and Si-O bonds in the coating. Compare the C-O bond strength in Fig.6
290 (b), the binding energy of 532.58 eV is mainly attributed to the Si-O bond (SiO₂)
291 caused by SiO₂ coating on FePc surface. About FePc@SiO₂-M (Fig.6d), the binding
292 energy of Si-O in the O1s fitting peak slightly changes because of the existing of C-O
293 and Si-O bonds in KH602.

294 The binding energy of Si2p (102.77 eV) is less than that of SiO₂ gel which is
295 reported to be 103.5 eV. The change of Si2p binding energy implies that after coating
296 with SiO₂, there may be a large number of Fe–O–Si bonds on FePc surface. Base on
297 the fitting of Si2p of FePc@SiO₂ (in Fig.6e), there are 2 peaks at binding energy of

298 101.58 eV and 102.86 eV. The binding energies of 102.86 eV and 101.58 eV
299 respectively represent Si–O bond in SiO_2 and Fe–O–Si bond in the interface of layer
300 [33]. Fig.6(f) shows the curve fitting of Si2p of $\text{FePc}@\text{SiO}_2\text{-M}$. It follows that Si2p
301 has three peaks at 100.25 eV, 101.89 eV and 102.81 eV, corresponding to Si–C in
302 KH602, Si–O–Fe and Si–O in coated FePc, respectively [39].

303 *3.5 Thermal analysis*

304 The thermogravimetric curves of FeOOH, and FePc before and after coating are
305 shown Fig.7(a). The thermal behaviour of FeOOH can be divided into two stages
306 during the entire heating process. One is the removal of adsorbed water on surface of
307 FeOOH within the range of 100 °C–270 °C, and the other is the weightlessness
308 between 270°C and 340°C, which is caused by the removal of crystalline water on the
309 surface of FeOOH [40], and the total mass loss rate reaches 10.05%. Comparing the
310 decomposition of FeOOH, the weight loss of FePc is significantly faster between 350
311 °C and 400 °C, and the maximum weight loss temperature is 373 °C, which is mainly
312 due to the decomposition of CuPc. The TG curve of $\text{FePc}@\text{SiO}_2$ is similar to that of
313 FePc, and the mass loss rate of $\text{FePc}@\text{SiO}_2$ in the whole process decreases to 8.23%
314 and 10.21% respectively, implying that the mass loss rate of coated FePc decreases
315 owing to the outstanding heat resistance of coating layer of amorphous silica.
316 Moreover, The mass loss rate of $\text{FePc}@\text{SiO}_2\text{-M}$ is 8.49% and 12.22% respectively.
317 This is because of the presence of KH602 on $\text{FePc}@\text{SiO}_2$ surface, the alkyl chain
318 structure of KH602 is susceptible to thermal decompose, as a result the mass loss rate
319 increases compared with $\text{FePc}@\text{SiO}_2$.

320 **Insert Fig. 7**

321 There is a mass loss peak in the range of 100-200 °C, compared with DTG curves
322 (Fig.7b), implying that FeOOH gradually dehydrates in this range. The peak of mass

323 loss changes significantly between 250 °C and 350 °C. The mass loss peak position of
324 FePc, FePc@SiO₂ and FePc@SiO₂-M all moves to the high temperature, indicating
325 that their thermal decomposition temperature increases, compared with that of
326 FeOOH. The SiO₂ forms a dense coating layer on the surface of FePc, which hinders
327 heat transfer during heating. In addition, after coating and surface treatment, the mass
328 loss peak of FePc is sharper than that of FePc@SiO₂ and FePc@SiO₂-M, that is the
329 decomposition of FePc is fastest within the same range.

330 Compared with the differential thermal analysis (DTA) curves of Fig.7c and Fig.7d,
331 the temperature of FeOOH endothermic peak is 306 °C, and that of FePc, FePc@SiO₂
332 and FePc@SiO₂-M are 309 °C, 310 °C and 314 °C, respectively. In particular, the
333 maximum thermal decomposition temperature of FePc@SiO₂-M is 435 °C (7c). The
334 results indicate that when FeOOH combines with CuPc to form FePC, its maximum
335 decomposition temperature and heat resistance are significantly improved. Combined
336 with the analysis of TEM, we believe that the function of SiO₂ coated on FePc surface
337 is to improve the heat resistance of FePc effectively. Moreover, the temperature of
338 endothermic peak about FePc@SiO₂-M in DTA curve is 314°C. Further more the
339 decomposition temperature increases slightly compared with that of FePc@SiO₂,
340 which is attributable to the presence of KH602 on the surface of FePc@SiO₂. During
341 the heating process, the KH602 is first heated and carbonized, and attached on
342 FePc@SiO₂ surface. In the subsequent heating process, the formed carbonization
343 layer blocks the heat transfer to FePc@SiO₂, so the thermal decomposition
344 temperature of FePc@SiO₂ increases. The above results strongly show that the
345 surface coated and modification successfully improves the heat resistance of FePc
346 pigment, the prepared green pigments are also expected to be used in the
347 high-temperature coating industry.

348 3.6 Hydrophobicity and dispersibility test

The water contact angle (WAC) test was used to evaluate the hydrophilicity/hydrophobicity of the pigment surface, and the results are shown in Fig.8A. The WAC of FeOOH is 7.79° , that is FeOOH has high hydrophilicity. This is due to the presence of -OH groups on the surface of FeOOH, which makes the surface of FeOOH hydrophilic. The WAC of CuPc is 84.19° , that is because the molecular structure of CuPc is an organic macrocyclic structure composed of C, O and N elements. The WAC of FePc is 86.38° , and the hydrophobic property is slightly increased, compared with that of CuPc. The WAC of FePc@SiO₂ is reduced to 75.68° , which is due to the existence of a lot of -OH groups on the surface of SiO₂ coating. When SiO₂ is coated on the surface of FePc, the hydroxyl number of the FePc@SiO₂ increases, so the WAC decreases slightly. After modification with KH602, the WAC of FePc@SiO₂-M increases to 91.14° , showing hydrophobic performance. This is due to the carbon chain structure of the KH602. Although there is no obvious absorption peak of alkyl chain of KH602 in FT-IR spectrum, the change of WAC of FePc@SiO₂-M also indirectly confirmed that the FePc is coated with SiO₂, and modified with KH602, successfully.

Insert Fig. 8

In order to further analyze the dispersion effect of FePc@SiO₂-M, industrial grade samples iron green pigment (835 type) were used for dispersion comparison experiments. Weigh 50 mg of sample and place it in 20 mL of water or chloroform solution. After ultrasonic dispersion for 1 min, let it stand for 24h, and compare the dispersion in different solvents. The results are shown in Fig.8B. Samples (a) and (b) show the dispersion of 835 type and FePc@SiO₂-M in water, and samples (c) and (d) show the dispersion of 835 type and FePc@SiO₂-M in chloroform, respectively.

373 Comparing the photos, it is found that in aqueous solution and chloroform solvent, the
374 dispersity of FePc@SiO₂-M is better than that of industrial 835 type. This may be due
375 to the presence of a large number of hydroxyl groups on the surface of FePc@SiO₂-M
376 after coating , which is conducive to improving dispersion in aqueous solutions. After
377 modification with KH602, there are carbon chains on the surface of FePc@SiO₂-M,
378 which is conducive to dispersion in organic solvents. Unlike inorganic green pigments,
379 the prepared green pigments are expected to be used in fields such as plastics and
380 coatings.

381 *3.7 Color Difference Analysis*

382 The color difference parameters and CIE coordinate diagram of coated and
383 modified pigments are shown in the Table 1 and Fig.9. When the ratio of CuPc to
384 FeOOH is 1:5, the *L** value is 55.44, and the *a** value reaches -9.9 (negative).
385 Comparing the *a** values of pigments, it is found that although the *a** values of CuPc
386 and FeOOH are greater than 0, after the above pigments are compounded, the *a**
387 values of FePc is negative and the pigments are green. Furthermore, comparing the
388 value of *b** , it is found that the *b** value of FeOOH is greater than 0, indicating
389 FeOOH has yellow phase, while the *b** value of CuPc is less than 0, indicating CuPc
390 has blue phase. After the combination of two pigments, the color *b** value was still
391 greater than 0, and there is still an obvious yellow component. About FePc@SiO₂
392 pigment, the value of *L** and *b** decreased slightly, while the *a** value increased
393 slightly but remained negative (-8.5). The results show that the brightness of the
394 FePc@SiO₂ slightly decreased, the blue tone increased, and it is still green tone but
395 the green color faded. This is because the SiO₂ coating layer is white, and the
396 presence of the coating layer reduces the brightness and green tone of the pigment.
397 The value of *L** and *b** of FePc@SiO₂-M are similar to that of FePc@SiO₂, but the

398 value of a^* increases. The color coordinate Hab^0 of the pigment changed from 166.72
399 to 159.21, indicating that the hue angles of the FePc@SiO₂-M is also lie in the green
400 region of the cylindrical color space. This indicates that after surface modification
401 with KH602, the brightness and yellow-blue tone of the FePc@SiO₂-M unchanged,
402 while the green tone slightly fades. The experimental results show that the KH602 has
403 little effect on the color of the FePc@SiO₂ pigment.

404 **Insert Table 1**

405 To quantify the differences in the color perceived by the human eye, color
406 coordinates were calculated for the pigments, as shown in Fig.9. In the CIE
407 chromaticity diagram (Fig. 9), point a (FeOOH) falls in the yellow area, while point b
408 (CuPc) appears blue. The pigments (c-e) fall into the yellow-green region in the CIE
409 coordinate, indicating that the color of the prepared green pigments is green tone. the
410 color coordinates of FePc and FePc@SiO₂ are coincide, the color coordinate of
411 FePc@SiO₂-M is biased towards the yellow-green region, but close to that of
412 FePc@SiO₂.

413 **Insert Fig. 9**

414 *3.8 Color stability*

415 In order to evaluate the heat resistance of FePc@SiO₂-M, the color stability of
416 samples before and after baking was measured. The results shown in Table 2. The
417 value of L^* slightly decreased, the a^* and b^* values slightly increased. According to
418 the color difference parameters of FePc@SiO₂-M before and after baking, the
419 calculated value of color difference ΔE is 1.623, and less than 5, which can meet the
420 industrial requirements for heat resistance of pigments. The experiment shows that
421 before and after baking, The color difference of FePc@SiO₂-M is not obvious, that is
422 FePc@SiO₂-M has excellent heat resistance and color stability.

423

Insert Table 2

424 The structure of the coating layer is closely related to the color change of the
425 pigment after being heated. The combination of FeOOH and CuPc reduces the a^*
426 value and the pigment appears green. The presence of CuPc improves the heat
427 resistance of the pigments due to the initial decomposition temperature of CuPc being
428 430 °C. When coated with SiO₂, the a^* value slightly increases, and the pigments
429 have a green tone. The coating layer SiO₂ inhibits dehydration, further improving the
430 heat resistance of the pigment. After surface modification with KH602, the a^* value
431 almost unchanged, and the pigment remains green tone. The coupling agent is
432 carbonized by heat during the high-temperature process, blocking heat transfer, and
433 effectively improving the heat resistance of the pigment. The prepared green pigment
434 overcomes the disadvantage of FeOOH being prone to color change at high
435 temperatures, and can maintain a green tone.

436 *3.9 Optical performance*

437 The minimum absorption peak of CuPc is 455 nm (blue light region) (Fig.10(a)),
438 the chromophore conjugation system of phthalocyanine is a polycyclic molecule
439 composed of four indoles, which contains a cyclic annulene with 18 electrons. In
440 addition, the minimum absorption peak of FeOOH is 583 nm (yellow light region),
441 but the minimum absorption peak of FePc is 537 nm. When these two pigments are
442 combined, according to the subtractive color-mixing principle [41], FeOOH absorbs
443 light near 350-450 nm that CuPc does not absorb; and CuPc absorbs light near
444 550-700 nm that the FeOOH does not absorb. The remaining light is only around 450
445 -550 nm, which is not completely absorbed by yellow and blue pigments. At the last,
446 this part of the light is reflected, and when we see it, it is a turquoise light. In addition,
447 the absorption peak position and intensity of the visible light absorption curve of the

448 composite pigment at about 667 nm have certain changes, compared to the visible
449 light absorption curve of CuPc. This indicates that the interaction between FeOOH
450 and CuPc occurs at the interface, which changes the coordination field splitting
451 energy.

452 **Insert Fig. 10**

453 The difuse reflectance of prepared pigments at 300-2500 nm is depicted in
454 Fig.10(b). The reflectivity of FeOOH is high than that of CuPc, while the reflectivity
455 of composite pigments is between the above two pigments. After weighting treatment
456 by multiplying the experimental spectral reflectance with the normalized spectral
457 irradiance of the sun, the NIR reflectance (R) was obtained. The FeOOH exhibited a
458 NIR reflectance of 64.3%, while the NIR reflectance of CuPc is 53.9%. The
459 composite pigments FePc exhibited an enhanced NIR reflectance, which is ascribed to
460 the FeOOH. The high NIR reflectance (64.5%) of FePc@SiO₂ could be ascribed to
461 the multiple refection of lights on thin layers of SiO₂. However, the KH602 slight
462 decreased the NIR reflectance of pigment to 57.9%. Compared to the maximum NIR
463 reflectance of pure Cr₂O₃ (49.0%) [42], all the prepared green pigments show high
464 NIR reflectance. More importantly, all the prepared green pigments exhibit high NIR
465 reflectance, which lets them suitable for use as cooling pigments.

466

467 **3 Conclusion**

468 Iron oxide green pigments, a composite green pigments of phthalocyanine blue and
469 FeOOH pigments, was prepared by intermittent stirring. The characterization results
470 shown that the aspect ratio of FePc decreased compared to FeOOH pigment. XPS
471 analysis shown that CuPc and FeOOH pigments were bound by hydrogen bonds. The
472 thermal decomposition temperature of the coated pigments increased from 266 °C to

473 314 °C, significantly improving its heat resistance. In addition, the color difference ΔE
474 of pigment was 1.623 after baking at 240 °C for 30 min. The surface modification of
475 composite pigments improved their hydrophobicity and dispersibility in organic
476 media. After the combination of FeOOH and CuPc pigments, the a^* value of the
477 pigments changed from a positive value to a negative value, presenting the pigments
478 had green tone. The coating and surface modification had improved heat resistance to
479 some extent, but had little impact on the a^* value of the pigment, and the pigment
480 remains in a green tone. Particularly, all the newly designed green pigments displayed
481 high NIR reflectance. The prepared pigments have high heat resistance and dispersion
482 properties, and are expected to be used in fields such as plastics and high-temperature
483 coatings. More noteworthy is that the coated green pigments can also be used as cold
484 pigments.

485

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490

491 CRediT authorship contribution statement

492 **Minhong Xu** designed experiments and wrote the main manuscript text, **Yuhua Guo**
493 analyzed sequencing data, **Guoxiang Pan** (Corresponding Author) wrote - review the
494 manuscripts, **Qiang Shen** analyzed sequencing data, **Mengyu Zhou** analyzed
495 experimental results, **Qiu Liang** carried out experiments. All authors reviewed the
496 manuscript.

497

498 **Declarations**

499 **Conflict of interest** The authors declare that they have no conflict of interest.

500

501 **References**

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