

# **The color rendering and near infrared reflection properties of coated iron oxide green pigments**

Minhong Xu, Yuhua Guo, Guoxiang Pan\*, Qiang Shen, Mengyu Zhou, Qiu Liang

Department of Materials Engineering, Huzhou University, Huzhou 313000, P.R. China

## **Abstract**

In this study, a novel iron oxide green pigment (FePc), composed of phthalocyanine blue (CuPc) and iron oxide yellow pigments (FeOOH), was prepared by intermittent stirring. The FePc was coated with the hydrolysate of sodium metasilicate and silane coupling agent KH602. The structure and properties of prepared pigments were characterized by TEM, SEM, XRD, FT-IR, XPS, and TG-DTA techniques. The aspect ratio of FePc decreased compared to FeOOH. XPS analysis showed that CuPc and FeOOH were bound by hydrogen bonds. The thermal decomposition temperature of the coated pigments increased from 266 °C to 314 °C, and the color difference  $\Delta E$  of pigment was 1.623 after baking at 240 °C for 30 min, indicating the prepared pigments had excellent heat resistance. The surface modification of FePc improved their hydrophobicity and dispersibility in organic media. The pigments with CuPc/FeOOH ratio of 0.2 exhibited well color properties ( $L^*=55.44$ ,  $a^*=-9.9$ ,  $b^*=6.63$ ), and green tone. The coating and surface modification had little impact on  $a^*$  value. Particularly, all the newly prepared green pigments displayed high NIR reflectance. The prepared pigments have high heat resistance and dispersion properties, and are expected to be used in fields such as plastics and high-temperature coatings.

---

\* Received

Corresponding author, Guoxiang Pan, Email: pgxzjut@163.com

**Keywords:** iron oxide yellow pigment, phthalocyanine, green pigment, coated, heat resistance

## 1. Introduction

The existence of natural dyes and pigments makes the world rich and colorful. However, natural dyes or pigments extracted from plants and animals exhibit a lack of stable color tone, poor chemical durability and thermal stability when subjected to heating, light, and chemical substances [1]. As a result, people have also attempted to synthesize pigments with different colors through various methods. Various shades of green pigment are reported such as uvarovite ( $\text{Ca}_3\text{Cr}_2(\text{SiO}_4)_3$ ) [2], Cr-doped  $\text{BiPO}_4$  [3],  $\text{Cr}_2\text{O}_3/\text{SiO}_2$  composite pigments [4]. These green pigments contain chromium ions. More importantly, chromium ions exhibit five valence states from +2 to +6, which transition to divalent or trivalent cations as the environment changes, ultimately leading to uncertainty in the color of the pigment [5]. In addition, among various oxides, chromium has multiple oxidation states, and Cr single bonds and O valence bonds can change under changes in temperature and/or oxygen partial pressure, thereby affecting the color tone of Cr based pigments. In actual production, the color tone of Cr based pigments is also difficult to control [6]. Other green pigments such as Scheele's green ( $\text{CuHAsO}_3$ ) and Emerald green ( $\text{Cu}(\text{CH}_3\text{CO}_2)\text{AsO}_3$ ), despite their high light-fastness, excellent price-performance ratio, brilliant hue, and high thermal stabilities [7], contain arsenic, making them undesirable. More importantly, inorganic green pigments are difficult to use in the high-temperature paint coating field. In addition, organic green pigments mainly consist of phthalocyanine green, which has well light and weather resistance, strong coloring power, bright color, small color difference, good color spreading and fluidity [8, 9]. However, the price of phthalocyanine green pigment is relatively high. Therefore, there is an urgent need to

develop environmentally friendly and cost-effective green pigments.

Preparing green pigments through color blending is also a new approach. Through the subtractive method in the principle of color mixing, it is found that green pigments can be prepared by blending yellow and blue pigments. Commonly used yellow pigments include doped  $\text{ZrO}_2$ ,  $(\text{Bi,Ca,Zn})\text{VO}_4$ ,  $\text{BiVO}_4\text{-GdPO}_4$ , Ce substituted  $\text{SrMoO}_4$  [10-13]. Goethite ( $\alpha\text{-FeOOH}$ ) is a common and stable iron oxide present in natural environment [14]. So far, due to the advantages of chemical stability, non-toxic and low cost,  $\alpha\text{-FeOOH}$  has been extensively studied in water purification, organic pollutant degradation, coatings, pigments and other fields [15-18]. Regarding blue pigments, there are Han blue ( $\text{BaCuSi}_4\text{O}_{10}$ ) and Maya blue used a long time ago [19], as well as Cerulean blue ( $\text{CoSnO}_3$ ) and Prussian blue ( $\text{Fe}_4[\text{Fe}(\text{CN})_6]_3$ ) used in currently [20-22]. Another type of blue pigment is organic blue pigment, such as phthalocyanine blue. Phthalocyanine blue is an important organic pigment with bright colors and strong coloring power [23, 24]. The preparation of green pigments using inorganic iron oxide yellow pigment and organic blue pigment phthalocyanine blue as raw materials is very challenging. It is necessary to consider the combination between organic and inorganic pigments and the color rendering performance.

It's important to note that ordinary iron oxide yellow pigment  $\alpha\text{-FeOOH}$  begin to dehydrate at 177 °C and continue to dehydrate to become  $\text{Fe}_2\text{O}_3$  between 270 °C and 320 °C, thus changing the color of pigment and affecting its application in high temperature occasions [25, 26]. Therefore, it is necessary to improve the heat resistance of the new green pigments synthesized from  $\alpha\text{-FeOOH}$ , so that it can meet the requirements of the coating of the coil material with the heat-resistant temperature between 210 and 250 °C. On the other hand, it is also necessary to improve the compatibility of synthesized green pigments with organic matrix and the wettability

of the matrix in order to broaden the application of synthesized green pigments in high temperature coating industry such as plastic processing and coil coatings. Recently, the focus of pigments formulators has shifted to develop materials with high near-infrared reflectivity (NIR) while maintaining their appropriate spectroscopic features that impart colour in the visible spectrum [27, 28]. The NIR reflective pigments have significant advantages in thermal protection of containers, ceiling coatings, and storage room surfaces to reduce the use of air conditioning.

In this work, inorganic FeOOH was used as the core, organic phthalocyanine blue pigment was coated externally to prepare composite green pigment (FePc). The FePc was modified to improve its heat resistance, and finally surface modification was carried out with a coupling agent to improve the dispersion performance of pigments in organic media. Besides, the prepared pigments were characterized by multifarious tools to study crystal structure, morphological characteristics, color coordinate values and optical performance. In addition, studied the heat resistance, coloration, and dispersion properties of pigments to evaluate their application. More importantly, the coloration mechanism and near-infrared reflectance performance based on spectral analysis have been further investigated.

## **2. Experimental Section**

### **2.1 Materials**

Phthalocyanine blue pigments ( CuPc, C.I.pigment blue 15: 3), were purchased from Shandong Yousuo Chemical Technology Co., Ltd. Iron oxide yellow pigments (FeOOH), were provided by Zhejiang Huayuan Pigments Co., Ltd. Sodium silicate, dilute sulfuric acid, silane coupling agent KH602, etc. were analytical purity and were purchased from East China Pharmaceutical Co., Ltd. All solutions were prepared with

99 distilled water.

## 100 2.2 Preparation of coated iron oxide green pigments

101 Weigh the pigments according to the mass ratio of FeOOH to CuPc was 5:1. The  
102 pigment was put into a high-speed pulverizer for mixing. After mixing for 1min, stop  
103 for 1 min, and then continue mixing. The final mixing time was 10 min. The iron  
104 oxide green pigment was obtained and marked as FePc.

105 The preparation of coated FePc was follows: FePc (10g) was added in 350 mL  
106 distilled water, then dispersing for 10 min with ultrasound to get suspension. Under  
107 water bath at 70 °C, the aboved suspension was stirred at 300-400 rpm. Certain  
108 quality of sodium silicate solid, which was calculated by the mass of silicon dioxide  
109 and the mass was 8% of the mass of FePc, was put into another beaker, and 200 mL  
110 of water was added to make it completely dissolved. Then the above solution was  
111 placed in a constant pressure funnel and dropped into the FePc suspension at a  
112 constant rate of 1 d/s, with a dripping time of about 60 min. After dropping, adjust the  
113 pH value of the solution to 8 with sulfuric acid, and continue stirring for 0.5 h. During  
114 this process, the pH value remained unchanged. Finally, the precipitate was filtered,  
115 washed with distilled water for three times and then dried at 70°C for 15 h. After that,  
116 the above product was identified as FePc@SiO<sub>2</sub>.

117 Coated FePc@SiO<sub>2</sub> was modified with silane coupling agents KH602. 5g  
118 FePc@SiO<sub>2</sub> was added in 250 mL distilled water, then dispersing for 10 min to get  
119 suspension. In water bath 80 °C, the aboved suspension was stirred at 300-400 rpm.  
120 The KH602 (8%, based on the quality of FePc@SiO<sub>2</sub>) was dissolved in 50mL  
121 distilled water, slowly dropped into FePc@SiO<sub>2</sub> suspension, and stirred for 90 min.  
122 Finally, the precipitate was filtered, washed with distilled water for three times and  
123 then dried at 70°C for 15 h. After that, the above product was marked as

FePc@SiO<sub>2</sub>-M.

### 2.3 Characterizations

Scanning electron microscopy (SEM) analysis was performed on a Hitachi Model S-3400N electron microscope operated at an accelerating voltage of 15 kV. Transmission electron microscopy (TEM) images were taken on a FEI Tecnai G2 F20 electron microscope. X-ray diffractometer (Cu K $\alpha$ ,  $\lambda$ =0.1545 nm, pipe voltage 36 kV, pipe current 20 mA) (Beijing Purkinje General Instrument Co., Ltd., China). Infrared spectra were recorded using the KBr disc method on a 5700 FT-IR spectrophotometer (Nicolet, USA). X-ray photoelectron spectroscopy (XPS) spectra of the solids were collected using a 250XI K-alpha spectrometer (Thermo Fisher Scientific), with a monochromatic Al K $\alpha$  radiation source and a hemi-spherical electron analyzer. Survey scans were recorded with a 500  $\mu$ m spot size and a fixed energy pass of 100 eV, where energy step size was 1.000 eV. The thermal properties were measured by simultaneous thermal analyzer SDT Q600 (TA instruments, USA), heating from the room temperature up to 400 °C with a heating rate of 10 °C/min. The wettability of sample is evaluated by SZ-CAM series contact angle tester (Xuanzhun instrument). Water droplets were deposited directly on the surface of the material and the water contact angles (WCA) were measured. Three measurements were performed per sample and averaged. The volume of the water droplet was 5  $\mu$ L and the tip used was a precision stainless steel tip.

### 2.4 Color and UV-Vis-NIR measurements

The visible light absorption spectrum and NIR region (700–2500 nm) was recorded by the UV-Vis-NIR spectrophotometer (Lambda 1050+, Perkin-Elmer). The CIE LAB color data ( $L^*$ ,  $a^*$ , and  $b^*$ ) were calculated from the visible light reflection spectrum by color CIE software (Perkin-Elmer, CIE D65 photo source, and 10

observation angle; the calculated spectrum range was 400–700 nm)[30]. The NIR reflectance (R) in the wavelength range of 700 nm to 2500 nm can be calculated by the following formula:

$$R = \frac{\int_{700}^{2500} r(\lambda)i(\lambda)d\lambda}{\int_{700}^{2500} i(\lambda)d\lambda} \quad (1)$$

where  $r(\lambda)$  is the spectral reflectance ( $\text{W}\cdot\text{m}^{-2}$ ) obtained from the UV–Vis–NIR spectrophotometer and  $i(\lambda)$  is the solar spectral irradiance ( $\text{W}\cdot\text{m}^{-2}\cdot\text{nm}^{-1}$ ) obtained from ASTM standard G159-98 [29]. NIR reflectance value was used to quantify the solar radiation, evaluate their reflective performance.

Calculation of the color difference ( $\Delta E$ ) after baking was used to evaluate the color stability of pigments. The sample (2.0 g) was put into crucible, and baked at 240 °C for 30 min. Then taken out and cooled to room temperature to measure the value of color difference. According to Hunter's color difference formula [31]:

$$\Delta E = (\Delta L^2 + \Delta a^2 + \Delta b^2)^{1/2} \quad (2)$$

Usually,  $\Delta E$  is the value of color difference,  $\Delta L$  is the lightness axis (positive is for white and negative is for black). The parameters  $\Delta a$  (positive is for red and negative is for green) and  $\Delta b$  (positive is for yellow and negative is for blue) indicate the hue or color of the pigments.

### 3. Results and discussion

#### 3.1 Morphology analysis

##### Insert Fig. 1

SEM images of FeOOH and the prepared FePc are displayed in Fig.1. The color of FeOOH (Fig.1a) is obvious yellow, and it has a typical needle-like structure with a length of about 2  $\mu\text{m}$ , and a small amount of agglomeration between particles. When

CuPc and FeOOH are blended to prepare FePc, the FePc pigment shows obvious green. The FePc are mostly rod-shaped, with a reduced aspect ratio and a reduced agglomeration phenomenon. Comparing the morphology, the particle size and length of FePc decreased, which may be caused by high speed stirring and shearing during the preparation process.

Fig. 1c shows the multi-elemental EDS mapping images of C, N, O, Fe, and Cu in FePc. The O and Fe elements come from FeOOH, the N element mainly comes from N in the molecular structure of CuPc. Apparently, the corresponding N and Cu maps illustrate a homogeneous distribution of these elements in the field of the cross-section view. At the same time, it is also certified that the uniform distribution of CuPc in iron oxide green pigment.

To further confirm the formation of the FePc@SiO<sub>2</sub> interface before and after modification, the TEM analysis were performed. Fig. 2(a) is the TEM diagram of FePc coated with silica. The surface of FePc is uniformly coated with a thin coating layer, and the thickness between 3.2 and 4.3 nm. This layer may be SiO<sub>2</sub>, indicating that SiO<sub>2</sub> has been uniformly coated on FePc surface.

The lattice fringes of FeOOH are clearly visible, and the spacing of (110) planes is calculated to be 0.42 nm (Fig. 2b). From Fig.2c and Fig.2d, the KH602 is modified successfully on FePc@SiO<sub>2</sub> surface, the coating layer is uniform, and the thickness of coating layer is about 6 nm. In conclusion, the coating modification of FePc is successfully realized using SiO<sub>2</sub> and KH602.

## Insert Fig. 2

### 3.2 XRD Characterization

## Insert Fig. 3

The XRD patterns of FePc, FePc@SiO<sub>2</sub> and FePc@SiO<sub>2</sub>-M are shown in Fig.3.



The diffraction peaks at about  $2\theta=17.8^\circ$ ,  $21.2^\circ$ ,  $34.7^\circ$ ,  $35.5^\circ$ ,  $36.6^\circ$ ,  $53.1^\circ$  and  $59.0^\circ$  are correspond to crystal planes (020), (110), (021), (101), (111), (221) and (151), respectively, which are consistent with the characteristic diffraction peaks and crystal planes of FeOOH[32]. Using the Bragg's formula ( $2d\sin\theta = n\lambda$ ) to calculate the interlayer space  $d_{110}$  value of FeOOH, the  $d_{110}$  value is 0.4199nm. The diffraction peaks of FePc and FeOOH are basically the same, and there are strong diffraction peaks near  $22^\circ(2\theta)$ ,  $33^\circ(2\theta)$  and  $37^\circ(2\theta)$ , with sharp peak patterns. This shows that FePc has well crystal form and high crystallinity, which means that the original crystal structure of FeOOH is not changed after compounding with CuPc.

Compared with the diffraction peak of FePc, there is no obvious characteristic peak of  $\text{SiO}_2$  in  $\text{FePc@SiO}_2$ , which means that coating layer  $\text{SiO}_2$  exists in amorphous state [33]. Furthermore, the positions of diffraction peaks of  $\text{FePc@SiO}_2\text{-M}$  are basically the same as that of  $\text{FePc@SiO}_2$ . The crystal structure of FePc did not change during the coating and surface modification processes.

### 3.3 FT-IR Characterization

#### Insert Fig. 4

The FTIR spectra of FeOOH,  $\text{FePc@SiO}_2$  and  $\text{FePc@SiO}_2\text{-M}$  are shown in Fig. 4. The absorption peaks (Fig.4a) of FeOOH at  $3420\text{ cm}^{-1}$  are owing to surface adsorbed water, and the strong absorption peak at  $3125\text{ cm}^{-1}$  is attributed to the stretching vibration of  $-\text{OH}$  in FeOOH. Four absorption peaks near  $899$ ,  $796$ ,  $615$  and  $475\text{ cm}^{-1}$  are characteristic absorption peaks of FeOOH [34]. The absorption peaks of CuPc at  $1723\text{ cm}^{-1}$  and  $1606\text{ cm}^{-1}$  are belong to the vibrational absorption peaks of  $\text{C}=\text{C}$  and  $\text{C}=\text{N}$  in the phthalocyanine ring [35]. By analyzing the infrared spectrum of FePc, it is found that the above two absorption peaks shift to  $1799$  and  $1663\text{ cm}^{-1}$ . This result may be due to the change of the electron cloud in the phthalocyanine ring, which is

caused by the physical and chemical interactions of the CuPc on FeOOH. Moreover, the absorption peak at about 3420  $\text{cm}^{-1}$  disappeared, but the absorption peak at 3125  $\text{cm}^{-1}$  does not change, indicating that there are still hydroxyl groups on FePc surface.

Compared with FePc@SiO<sub>2</sub> (Fig.4b.), the absorption peak position of FePc@SiO<sub>2</sub>-M at 3100  $\text{cm}^{-1}$  changed slightly. This may be due to the change of the hydroxyl group on the surface of FePc@SiO<sub>2</sub>, resulting in the change of the stretching vibration absorption peak of -OH group. However, the change of FT-IR spectrum before and after modification with KH602 is not obvious, which may be caused by less coating amount.

### 3.4 XPS analyses

The XPS spectra with full-scan of prepared pigments are shown in Fig.5a. The binding energies at about 724.7 eV, 711.5 eV, 529.7 eV, 398.5 eV, 285.7 eV, 154.15 eV, 103.5 eV, 91.1 eV and 52.7 eV are consistent with the electronic states of Fe2p, O1s, N1s, C1s, Si2s, Si2p, Fe3s and Fe3p, respectively [36]. The XPS spectrum of FeOOH shows three elements: O, C, and Fe. According to the chemical formula, only O and Fe belong to FeOOH, while C comes from excess hydrocarbons in the environment. The Auger resonance peak of O is in Fig.5a, but the area of the peak is not large, indicating that a small amount of O<sub>2</sub> is attached to FeOOH surface. The characteristic peaks of N1s, Cu2p<sub>1</sub> and Cu2p<sub>3</sub> in CuPc molecule appear in the XPS spectrum of FePc, and the intensity of the characteristic peaks of C1s increases. This is mainly due to the ring structure of C, N and the central metal Cu in CuPc. When FePc is coated with sodium silicate (FePc@SiO<sub>2</sub>), it can be clearly seen that the peaks of Si2s and Si2p are between 100-200 eV. When coated with KH602, the percentage of C and Si in FePc@SiO<sub>2</sub>-M decreases clearly, but the percentage of O significantly increases. This is probably related with the elements of KH602 and atomic

composition of the interface layer of FePc@SiO<sub>2</sub>-M.

The XPS spectrum of Cu2p (Fig.5b) shows that the binding energies of Cu2p<sub>3/2</sub> in FePc, FePc@SiO<sub>2</sub> and FePc@SiO<sub>2</sub>-M are 935.05 eV, 935.11 eV and 935.21 eV, respectively. It is found that the binding energy of Cu2p<sub>3/2</sub> increases slightly, indicating that the density of electron cloud around Cu atom increases, which is caused by the effect of SiO<sub>2</sub> and KH602. Furthermore, compared with the C1s binding energy of 285.7 eV, binding energy of C1s in FePc@SiO<sub>2</sub> (Fig.5c) is almost unchanged. It shows that the binding mode between CuPc and FeOOH remains unchanged after SiO<sub>2</sub> coated. After modification with HK602, the binding energy of C1s is changed from 285.7 eV to 285.4 eV, which is mainly attributed to the C-O in HK602. This indirectly confirms that the success of surface modification of FePc@SiO<sub>2</sub> with HK602.

The tested samples have N1s main nitrogen peak at about 398.6 eV (Fig.5c-e), which belongs to C-N bond. In addition, there is a weak  $\pi$ - $\pi^*$  satellite feature of the shift several electronvolt of nitrogen-containing aromatic substances from the main nitrogen peak near 400.3 eV [37]. The N1s binding energy in FePc@SiO<sub>2</sub> is 0.03 eV, which is higher than that of FePc, implying the intermolecular force increases with the increase of the electron cloud near N1s. The N element mainly comes from CuPc, indicating that there is intermolecular force between the coating layer and CuPc after coating. In addition, the binding energy of N1s about FePc@SiO<sub>2</sub>-M continues to increase (398.72 eV), which is due to the C-N bond in the molecular structure of KH602. The above experimental results show that both coating and surface modification can make the electron cloud near the N1s of FePc@SiO<sub>2</sub>-M increase, and the intermolecular force further increases.

### Insert Fig. 5

### Insert Fig. 6

As shown in Fig.6(a), about FeOOH, there are 3 peaks in the O1s fitting diagram, and their binding energies are 529.65 eV, 531.05 eV and 533.21 eV, respectively. Furthermore, the electron binding energy at 529.65 eV and 531.05 eV are ascribed to the O in OH of FeOOH and O in FeOOH, respectively, the results fit the feature structure of FeOOH [38]. The binding energy at 533.21 eV is from the C-O bond of air. Moreover, as shown in Fig.6(b), the O1s binding energy in FePc increases, compared with that in FeOOH, which shows the ability of oxygen to lose electrons in FePc is slightly reduced. According to the FT-IR analysis, the absorption peak of -OH in FePc does not shift. Therefore, the shift of binding energy of O1s may be because of the formation of hydrogen bond between O in FeOOH and H in the -C-H group of CuPc ring, resulting in the change of electron cloud near the C=C and C=N groups in phthalocyanine ring.

In Fig.6(c), there are also 3 peaks in the O1s fitting diagram in FePc@SiO<sub>2</sub>, and their binding energies are 529.9 eV, 531.21 eV and 532.58 eV, respectively. Also, these 3 binding energies are respectively relative to the O in FeOOH, O in FeOOH, and O in C-O and Si-O bonds in the coating. Compare the C-O bond strength in Fig.6 (b), the binding energy of 532.58 eV is mainly attributed to the Si-O bond (SiO<sub>2</sub>) caused by SiO<sub>2</sub> coating on FePc surface. About FePc@SiO<sub>2</sub>-M (Fig.6d), the binding energy of Si-O in the O1s fitting peak slightly changes because of the existing of C-O and Si-O bonds in KH602.

The binding energy of Si2p (102.77 eV) is less than that of SiO<sub>2</sub> gel which is reported to be 103.5 eV. The change of Si2p binding energy implies that after coating with SiO<sub>2</sub>, there may be a large number of Fe-O-Si bonds on FePc surface. Base on the fitting of Si2p of FePc@SiO<sub>2</sub> (in Fig.6e), there are 2 peaks at binding energy of

101.58 eV and 102.86 eV. The binding energies of 102.86 eV and 101.58 eV respectively represent Si–O bond in SiO<sub>2</sub> and Fe–O–Si bond in the interface of layer [33]. Fig.6(f) shows the curve fitting of Si2p of FePc@SiO<sub>2</sub>-M. It follows that Si2p has three peaks at 100.25 eV, 101.89 eV and 102.81 eV, corresponding to Si–C in KH602, Si–O–Fe and Si–O in coated FePc, respectively [39].

### 3.5 Thermal analysis

The thermogravimetric curves of FeOOH, and FePc before and after coating are shown Fig.7(a). The thermal behaviour of FeOOH can be divided into two stages during the entire heating process. One is the removal of adsorbed water on surface of FeOOH within the range of 100 °C–270 °C, and the other is the weightlessness between 270°C and 340°C, which is caused by the removal of crystalline water on the surface of FeOOH [40], and the total mass loss rate reaches 10.05%. Comparing the decomposition of FeOOH, the weight loss of FePc is significantly faster between 350 °C and 400 °C, and the maximum weight loss temperature is 373 °C, which is mainly due to the decomposition of CuPc. The TG curve of FePc@SiO<sub>2</sub> is similar to that of FePc, and the mass loss rate of FePc@SiO<sub>2</sub> in the whole process decreases to 8.23% and 10.21% respectively, implying that the mass loss rate of coated FePc decreases owing to the outstanding heat resistance of coating layer of amorphous silica. Moreover, The mass loss rate of FePc@SiO<sub>2</sub>-M is 8.49% and 12.22% respectively. This is because of the presence of KH602 on FePc@SiO<sub>2</sub> surface, the alkyl chain structure of KH602 is susceptible to thermal decompose, as a result the mass loss rate increases compared with FePc@SiO<sub>2</sub>.

#### Insert Fig. 7

There is a mass loss peak in the range of 100-200 °C, compared with DTG curves (Fig.7b), implying that FeOOH gradually dehydrates in this range. The peak of mass

loss changes significantly between 250 °C and 350 °C. The mass loss peak position of FePc, FePc@SiO<sub>2</sub> and FePc@SiO<sub>2</sub>-M all moves to the high temperature, indicating that their thermal decomposition temperature increases, compared with that of FeOOH. The SiO<sub>2</sub> forms a dense coating layer on the surface of FePc, which hinders heat transfer during heating. In addition, after coating and surface treatment, the mass loss peak of FePc is sharper than that of FePc@SiO<sub>2</sub> and FePc@SiO<sub>2</sub>-M, that is the decomposition of FePc is fastest within the same range.

Compared with the differential thermal analysis (DTA) curves of Fig.7c and Fig.7d, the temperature of FeOOH endothermic peak is 306 °C, and that of FePc, FePc@SiO<sub>2</sub> and FePc@SiO<sub>2</sub>-M are 309 °C, 310 °C and 314 °C, respectively. In particular, the maximum thermal decomposition temperature of FePc@SiO<sub>2</sub>-M is 435 °C (7c). The results indicate that when FeOOH combines with CuPc to form FePC, its maximum decomposition temperature and heat resistance are significantly improved. Combined with the analysis of TEM, we believe that the function of SiO<sub>2</sub> coated on FePc surface is to improve the heat resistance of FePc effectively. Moreover, the temperature of endothermic peak about FePc@SiO<sub>2</sub>-M in DTA curve is 314°C. Further more the decomposition temperature increases slightly compared with that of FePc@SiO<sub>2</sub>, which is attributable to the presence of KH602 on the surface of FePc@SiO<sub>2</sub>. During the heating process, the KH602 is first heated and carbonized, and attached on FePc@SiO<sub>2</sub> surface. In the subsequent heating process, the formed carbonization layer blocks the heat transfer to FePc@SiO<sub>2</sub>, so the thermal decomposition temperature of FePc@SiO<sub>2</sub> increases. The above results strongly show that the surface coated and modification successfully improves the heat resistance of FePc pigment, the prepared green pigments are also expected to be used in the high-temperature coating industry.

### 3.6 Hydrophobicity and dispersibility test

The water contact angle (WAC) test was used to evaluate the hydrophilicity/hydrophobicity of the pigment surface, and the results are shown in Fig.8A. The WAC of FeOOH is 7.79°, that is FeOOH has high hydrophilicity. This is due to the presence of -OH groups on the surface of FeOOH, which makes the surface of FeOOH hydrophilic. The WAC of CuPc is 84.19°, that is because the molecular structure of CuPc is an organic macrocyclic structure composed of C, O and N elements. The WAC of FePc is 86.38°, and the hydrophobic property is slightly increased, compared with that of CuPc. The WAC of FePc@SiO<sub>2</sub> is reduced to 75.68°, which is due to the existence of a lot of -OH groups on the surface of SiO<sub>2</sub> coating. When SiO<sub>2</sub> is coated on the surface of FePc, the hydroxyl number of the FePc@SiO<sub>2</sub> increases, so the WAC decreases slightly. After modification with KH602, the WAC of FePc@SiO<sub>2</sub>-M increases to 91.14°, showing hydrophobic performance. This is due to the carbon chain structure of the KH602. Although there is no obvious absorption peak of alkyl chain of KH602 in FT-IR spectrum, the change of WAC of FePc@SiO<sub>2</sub>-M also indirectly confirmed that the FePc is coated with SiO<sub>2</sub>, and modified with KH602, successfully.

#### Insert Fig. 8

In order to further analyze the dispersion effect of FePc@SiO<sub>2</sub>-M, industrial grade samples iron green pigment (835 type) were used for dispersion comparison experiments. Weigh 50 mg of sample and place it in 20 mL of water or chloroform solution. After ultrasonic dispersion for 1 min, let it stand for 24h, and compare the dispersion in different solvents. The results are shown in Fig.8B. Samples (a) and (b) show the dispersion of 835 type and FePc@SiO<sub>2</sub>-M in water, and samples (c) and (d) show the dispersion of 835 type and FePc@SiO<sub>2</sub>-M in chloroform, respectively.

Comparing the photos, it is found that in aqueous solution and chloroform solvent, the dispersity of FePc@SiO<sub>2</sub>-M is better than that of industrial 835 type. This may be due to the presence of a large number of hydroxyl groups on the surface of FePc@SiO<sub>2</sub>-M after coating, which is conducive to improving dispersion in aqueous solutions. After modification with KH602, there are carbon chains on the surface of FePc@SiO<sub>2</sub>-M, which is conducive to dispersion in organic solvents. Unlike inorganic green pigments, the prepared green pigments are expected to be used in fields such as plastics and coatings.

### 3.7 Color Difference Analysis

The color difference parameters and CIE coordinate diagram of coated and modified pigments are shown in the Table 1 and Fig.9. When the ratio of CuPc to FeOOH is 1:5, the  $L^*$  value is 55.44, and the  $a^*$  value reaches -9.9 (negative). Comparing the  $a^*$  values of pigments, it is found that although the  $a^*$  values of CuPc and FeOOH are greater than 0, after the above pigments are compounded, the  $a^*$  values of FePc is negative and the pigments are green. Furthermore, comparing the value of  $b^*$ , it is found that the  $b^*$  value of FeOOH is greater than 0, indicating FeOOH has yellow phase, while the  $b^*$  value of CuPc is less than 0, indicating CuPc has blue phase. After the combination of two pigments, the color  $b^*$  value was still greater than 0, and there is still an obvious yellow component. About FePc@SiO<sub>2</sub> pigment, the value of  $L^*$  and  $b^*$  decreased slightly, while the  $a^*$  value increased slightly but remained negative (-8.5). The results show that the brightness of the FePc@SiO<sub>2</sub> slightly decreased, the blue tone increased, and it is still green tone but the green color faded. This is because the SiO<sub>2</sub> coating layer is white, and the presence of the coating layer reduces the brightness and green tone of the pigment. The value of  $L^*$  and  $b^*$  of FePc@SiO<sub>2</sub>-M are similar to that of FePc@SiO<sub>2</sub>, but the



value of  $a^*$  increases. The color coordinate  $Hab^0$  of the pigment changed from 166.72 to 159.21, indicating that the hue angles of the FePc@SiO<sub>2</sub>-M is also lie in the green region of the cylindrical color space. This indicates that after surface modification with KH602, the brightness and yellow-blue tone of the FePc@SiO<sub>2</sub>-M unchanged, while the green tone slightly fades. The experimental results show that the KH602 has little effect on the color of the FePc@SiO<sub>2</sub> pigment.

#### Insert Table 1

To quantify the differences in the color perceived by the human eye, color coordinates were calculated for the pigments, as shown in Fig.9. In the CIE chromaticity diagram (Fig. 9), point a (FeOOH) falls in the yellow area, while point b (CuPc) appears blue. The pigments (c-e) fall into the yellow-green region in the CIE coordinate, indicating that the color of the prepared green pigments is green tone. the color coordinates of FePc and FePc@SiO<sub>2</sub> are coincide, the color coordinate of FePc@SiO<sub>2</sub>-M is biased towards the yellow-green region, but close to that of FePc@SiO<sub>2</sub>.

#### Insert Fig. 9

### 3.8 Color stability

In order to evaluate the heat resistance of FePc@SiO<sub>2</sub>-M, the color stability of samples before and after baking was measured. The results shown in Table 2. The value of  $L^*$  slightly decreased, the  $a^*$  and  $b^*$  values slightly increased. According to the color difference parameters of FePc@SiO<sub>2</sub>-M before and after baking, the calculated value of color difference  $\Delta E$  is 1.623, and less than 5, which can meet the industrial requirements for heat resistance of pigments. The experiment shows that before and after baking, The color difference of FePc@SiO<sub>2</sub>-M is not obvious, that is FePc@SiO<sub>2</sub>-M has excellent heat resistance and color stability.

## Insert Table 2

The structure of the coating layer is closely related to the color change of the pigment after being heated. The combination of FeOOH and CuPc reduces the  $a^*$  value and the pigment appears green. The presence of CuPc improves the heat resistance of the pigments due to the initial decomposition temperature of CuPc being 430 °C. When coated with SiO<sub>2</sub>, the  $a^*$  value slightly increases, and the pigments have a green tone. The coating layer SiO<sub>2</sub> inhibits dehydration, further improving the heat resistance of the pigment. After surface modification with KH602, the  $a^*$  value almost unchanged, and the pigment remains green tone. The coupling agent is carbonized by heat during the high-temperature process, blocking heat transfer, and effectively improving the heat resistance of the pigment. The prepared green pigment overcomes the disadvantage of FeOOH being prone to color change at high temperatures, and can maintain a green tone.

### 3.9 Optical performance

The minimum absorption peak of CuPc is 455 nm (blue light region) (Fig.10(a) ), the chromophore conjugation system of phthalocyanine is a polycyclic molecule composed of four indoles, which contains a cyclic annulene with 18 electrons. In addition, the minimum absorption peak of FeOOH is 583 nm (yellow light region), but the minimum absorption peak of FePc is 537 nm. When these two pigments are combined, according to the subtractive color-mixing principle [41], FeOOH absorbs light near 350-450 nm that CuPc does not absorb; and CuPc absorbs light near 550-700 nm that the FeOOH does not absorb. The remaining light is only around 450-550 nm, which is not completely absorbed by yellow and blue pigments. At the last, this part of the light is reflected, and when we see it, it is a turquoise light. In addition, the absorption peak position and intensity of the visible light absorption curve of the

composite pigment at about 667 nm have certain changes, compared to the visible light absorption curve of CuPc. This indicates that the interaction between FeOOH and CuPc occurs at the interface, which changes the coordination field splitting energy.

### Insert Fig. 10

The diffuse reflectance of prepared pigments at 300-2500 nm is depicted in Fig.10(b). The reflectivity of FeOOH is high than that of CuPc, while the reflectivity of composite pigments is between the above two pigments. After weighting treatment by multiplying the experimental spectral reflectance with the normalized spectral irradiance of the sun, the NIR reflectance (R) was obtained. The FeOOH exhibited a NIR reflectance of 64.3%, while the NIR reflectance of CuPc is 53.9%. The composite pigments FePc exhibited an enhanced NIR reflectance, which is ascribed to the FeOOH. The high NIR reflectance (64.5%) of FePc@SiO<sub>2</sub> could be ascribed to the multiple reflection of lights on thin layers of SiO<sub>2</sub>. However, the KH602 silight decreased the NIR reflectance of pigment to 57.9%. Compared to the maximum NIR reflectance of pure Cr<sub>2</sub>O<sub>3</sub> (49.0%) [42], all the prepared green pigments show high NIR reflectance. More importantly, all the prepared green pigments exhibit high NIR reflectance, which lets them suitable for use as cooling pigments.

### 3 Conclusion

Iron oxide green pigments, a composite green pigments of phthalocyanine blue and FeOOH pigments, was prepared by intermittent stirring. The characterization results shown that the aspect ratio of FePc decreased compared to FeOOH pigment. XPS analysis shown that CuPc and FeOOH pigments were bound by hydrogen bonds. The thermal decomposition temperature of the coated pigments increased from 266 °C to

314 °C, significantly improving its heat resistance. In addition, the color difference  $\Delta E$  of pigment was 1.623 after baking at 240 °C for 30 min. The surface modification of composite pigments improved their hydrophobicity and dispersibility in organic media. After the combination of FeOOH and CuPc pigments, the  $a^*$  value of the pigments changed from a positive value to a duplicate value, presenting the pigments had green tone. The coating and surface modification had improved heat resistance to some extent, but had little impact on the  $a^*$  value of the pigment, and the pigment remains in a green tone. Particularly, all the newly designed green pigments displayed high NIR reflectance. The prepared pigments have high heat resistance and dispersion properties, and are expected to be used in fields such as plastics and high-temperature coatings. More noteworthy is that the coated green pigments can also be used as cold pigments.

## Acknowledgments

The authors would like to acknowledge the financial support by Zhejiang Provincial Natural Science Foundation of China (LY21E040001) and "Pioneer" and "Leading Goose" R&D Program of Zhejiang (2023C01112).

## CRedit authorship contribution statement

**Minhong Xu** designed experiments and wrote the main manuscript text, **Yuhua Guo** analyzed sequencing data, **Guoxiang Pan** (Corresponding Author) wrote - review the manuscripts, **Qiang Shen** analyzed sequencing data, **Mengyu Zhou** analyzed experimental results, **Qiu Liang** carried out experiments. All authors reviewed the manuscript.

## Declarations

**Conflict of interest** The authors declare that they have no conflict of interest.

## References

- [1] S. Li, B. Mu, Z. Meng, Y. Zhu, H. Zhang, A. Wang, Facile color regulation of organic/inorganic hybrid pigments combining with palygorskite and CoNi layered double hydroxides, *Dyes Pigments*. 213 (2023) 111162. <https://doi.org/10.1016/j.dyepig.2023.111162>
- [2] B. Tang, Q. Xiang, Z. Fang, X. Zhang, Z. Xiong, H. Li, C. Yuan, S. Zhang, Influence of  $\text{Cr}^{3+}$  substitution for  $\text{Mg}^{2+}$  on the crystal structure and microwave dielectric properties of  $\text{CaMg}_{1-x}\text{Cr}_{2x/3}\text{Si}_2\text{O}_6$  ceramics, *Ceram Int.* 45 (2019) 11484-11490. <https://doi.org/10.1016/j.ceramint.2019.03.016>
- [3] C. Ding, A. Han, M. Ye, Y. Zhang, L. Yao, J. Yang, Synthesis and characterization of a series of new green solar heat-reflective pigments: Cr-doped  $\text{BiPO}_4$  and its effect on the aging resistance of PMMA (Poly(methyl methacrylate)), *Sol Energ. Mat. Sol. C*. 191 (2019) 427-436. <https://doi.org/10.1016/j.solmat.2018.12.007>
- [4] S. Ai, H. Zheng, J. Yu. Preparation and Reflectance Spectrum Modulation of  $\text{Cr}_2\text{O}_3$  Green Pigment by Solution Combustion Synthesis, *Materials*. 13 (2020) 1540. <https://doi.org/10.3390/ma13071540>
- [5] T. Chen, J. Zha, X. Zhang, X. Hu, W. Jiang, Z. Xie, J. Wan, Synthesis and characterization of  $\text{Pr}_x\text{Zr}_{1-x}\text{SiO}_4$  ( $x = 0-0.08$ ) yellow pigments via non-hydrolytic sol-gel method, *J. Eur. Ceram. Soc.* 38 (2018) 4568-4575. <https://doi.org/10.1016/j.jeurceramsoc.2018.05.021>

- [6] I. Mjejri, S. Mornet, M. Gaudon, From nano-structured polycrystalline spheres with  $\text{Zn}_{1-x}\text{Co}_x\text{O}$  composition to core-shell  $\text{Zn}_{1-x}\text{Co}_x\text{O}@\text{SiO}_2$  as green pigments, *J. Alloy Compound*. 777 (2019) 1204-1210.  
<https://doi.org/10.1016/j.jallcom.2018.10.333>
- [7] T. Thongkanluang, T. Kittiauchawal, P. Limsuwan, Preparation and characterization of  $\text{Cr}_2\text{O}_3\text{-TiO}_2\text{-Al}_2\text{O}_3\text{-V}_2\text{O}_5$  green pigment, *Ceram Int*. 37 (2011) 543-548.  
<https://doi.org/10.1016/j.ceramint.2010.09.044>
- [8] G. Germinario, I. Werf, L. Sabbatini. Pyrolysis gas chromatography mass spectrometry of two green phthalocyanine pigments and their identification in paint systems, *J. Anal. Appl. Pyrol*. 115 (2015) 175-183.  
<https://doi.org/10.1016/j.jaap.2015.07.016>
- [9] J. Choi, W. Lee, J.P. Kim, Broad-absorbing tetrabenzocorrolazine green photosensitizer for the enhanced conversion efficiency in dye-sensitized solar cells, *J. Porphyr. Phthalocya*. 20 (2016) 804-812.  
<https://doi.org/10.1142/S108842461650098X>
- [10] Z. Hu, Q. Wang, J. Liu, K. Liu, Q. Chang, Synthesis and chromatic properties of V-doped and V/Y-codoped  $\text{ZrO}_2$  yellow pigments, *J. Alloys Compd*. 856 (2021) 157397-157408.  
<https://doi.org/10.1016/j.jallcom.2020.157397>
- [11] T. Masui, T. Honda, Wendusu, N. Imanaka, Novel and environmentally friendly  $(\text{Bi,Ca,Zn})\text{VO}_4$  yellow pigments, *Dyes Pigments*. 99 (2013) 636-641.  
<https://doi.org/10.1016/j.dyepig.2013.06.030>
- [12] K.V.Raj Athira, P. Prabhakar Rao, K.Rajesh, Potential NIR reflecting yellow pigments powder in monoclinic scheelite type solid solutions:  $\text{BiVO}_4\text{-GdPO}_4$  for

cool roof applications, *Inorg. Chem. Commun.* 149 (2023) 110416.  
<https://doi.org/10.1016/j.inoche.2023.110416>

[13] V. Elakkiya, S. Sumathi, Low-temperature synthesis of environment-friendly cool yellow pigment: Ce substituted  $\text{SrMoO}_4$ , *Mater Lett.* 263 (2020) 127246-127250. <https://doi.org/10.1016/j.matlet.2019.127246>

[14] A. Mrm, B. Lz, A. Es, Synthesis and characterization of goethite ( $\alpha\text{-FeOOH}$ ) magnetic nanofluids, *Int. J. Thermofluids.* 15 (2022) 100169. <https://doi.org/10.1016/j.ijft.2022.100169>

[15] F. Fang, Y. Jia, P.Y. Wu, Q.Y. Zhang, Y.P. Jiang, S.S. Zhou, D.Y. Peng, Facile one-pot preparation of goethite/parabutlerite nanocomposites and their removal properties and mechanism toward As(V) ions. *Appl. Surf. Sci.* 324 (2015) 355-362. <https://doi.org/10.1016/j.apsusc.2014.10.174>

[16] X. Qian, M. Ren, Y. Zhu, D. Yue, Y. Han, J. Jia, Zhao Y., Visible Light Assisted Heterogeneous Fenton-Like Degradation of Organic Pollutant via  $\alpha\text{-FeOOH}$ /Mesoporous Carbon Composites, *Environ. Sci. Technol.* 51 (2017) 3993-4000. <https://doi.org/10.1021/acs.est.6b06429>

[17] M. cc. POMIS, M. Menu , C. Vignaud, Red Palaeolithic pigments: natural hematite or heated goethite? *Archaeometry*, 41 (1999) 275-285. <https://doi.org/10.1111/j.1475-4754.1999.tb00983.x>

[18] E. Sani, L. Mercatelli, M.R. Martina, S. Barison, F. Agresti, Magnetic-field tunability of optical properties in colloidal suspensions of goethite ( $\alpha\text{-FeOOH}$ ) nanorods, *Opt. Mater.* 96 (2019) 109303-109310. <https://doi.org/10.1016/j.optmat.2019.109303>

- [19] S.G. Menon, H.C. Swart, Microwave-assisted synthesis of blue-green  $\text{NiAl}_2\text{O}_4$  nanoparticle pigments with high near-infrared reflectance for indoor cooling, *J. Alloy Compd.* 81 (2020) 152991-152998.  
<https://doi.org/10.1016/j.jallcom.2019.152991>
- [20] S.G. Menon, D.N. Hebbar, S.D. Kulkarni, K.S. Choudhari, C. Santhosh, Facile synthesis and luminescence studies of nanocrystalline red emitting  $\text{Cr: ZnAl}_2\text{O}_4$  phosphor, *Mater. Res. Bull.* 86 (2017) 63-71.  
<https://doi.org/10.1016/j.materresbull.2016.10.009>
- [21] J.M. Rynkowski, T. Paryjczak, M. Lenik, On the nature of oxidic nickel phases in  $\text{NiO}/\gamma\text{-Al}_2\text{O}_3$  catalysts, *Appl. Catal. Gen.* 106 (1993) 73-82.  
[https://doi.org/10.1016/0926-860X\(93\)80156-K](https://doi.org/10.1016/0926-860X(93)80156-K)
- [22] X. Wang, F. Yu, X. Duan, D. Yuan, Effects of Annealing Temperature and  $\text{SiO}_2$  Matrix on the Structure and Optical Properties of Co-Doped  $\text{ZnAl}_2\text{O}_4/\text{SiO}_2$  Nanoglass–Ceramic Composites, *J. Phys. Chem. C.* 116 (2012) 2313-2321.  
<https://doi.org/10.1021/jp209837q>
- [23] H. Huang, Z. Lan, W. Li, W. Mo, J. Zhang, A novel and low-cost  $\text{CuPc}@C$  catalyst derived from the compounds of sunflower straw and copper phthalocyanine pigment for oxygen reduction reaction, *RSC Advances.* 11 (2021) 15590-15597.  
<https://doi.org/10.1039/D1RA01775F>
- [24] T. Zhang, C. Zhou, Properties of copper phthalocyanine blue (C.I. Pigment Blue 15:3) treated with poly(ethylene glycol)s, *Dyes Pigments.* 35 (1997) 123-130.  
[https://doi.org/10.1016/S0143-7208\(97\)84839-5](https://doi.org/10.1016/S0143-7208(97)84839-5)
- [25] I. Arčon, A. Kodre, M. Mozetič, XAS study of oxygen plasma-treated micronized iron oxide pigments, *Vacuum.* 80 (2005) 178-183.



- 598 <https://doi.org/10.1016/j.vacuum.2005.08.012>
- 599 [26] R. Snovski, J. Grinblat, M.T. Sougrati, J.C. Jumas, S. Margel, Synthesis and  
600 characterization of iron, iron oxide and iron carbide nanostructures, J. Magn.  
601 Magn. Mater. 349 (2013) 35-44.  
602 <https://doi.org/10.1016/j.jmmm.2013.08.043>
- 603 [27] R. Levinson, H. Akbari, P. Berdahl, Measuring solar reflectance—Part I:  
604 Defining a metric that accurately predicts solar heat gain, Sol. Energy. 84 (2010)  
605 1717-1744.  
606 <https://doi.org/10.1016/j.solener.2010.04.018>
- 607 [28] R. Levinson, H. Akbari, P. Berdahl, Measuring solar reflectance—Part II: Review  
608 of practical methods, Sol. Energy. 84 (2010) 1745-1759.  
609 <https://doi.org/10.1016/j.solener.2010.04.017>
- 610 [29] L. Yuan, X. Weng, M. Zhou, Q. Zhang, L. Deng, Structural and  
611 VisibleNearInfrared Optical Properties of Cr-Doped TiO<sub>2</sub> for Colored Cool  
612 Pigments, Nanoscale Res. Lett. 12 (2017) 597.  
613 <https://doi.org/10.1186/s11671-017-2365-5>
- 614 [30] Y. Li, S. Le, Z. Wang, Y. Hong, Q. Pu, Preparation and characterization of the  
615 Sr<sup>2+</sup>-doped  $\gamma$ -Ce<sub>2</sub>S<sub>3</sub>@c-SiO<sub>2</sub> red pigments exhibiting improved temperature and  
616 acid stability, Appl. Surf. Sci. 508 (2020) 145266.  
617 <https://doi.org/10.1016/j.apsusc.2020.145266>
- 618 [31] B. Becerir, Assessment of the results of different color difference formulae under  
619 different illuminants by wash fastness tests, Fiber Polym. 12 (2011) 946-956.  
620 <https://doi.org/10.1007/s12221-011-0946-y>
- 621 [32] M. Ohmori, E. Matijevi, Preparation and properties of uniform coated colloidal  
622 particles: 8. Silica on Iron, J. Colloid Interf. Sci. 160 (1993) 288-292.

- 623 <https://doi.org/10.1006/jcis.1993.1398>
- 624 [33] C. Yao, F. Wu, X. Lin, X. Yang, D. Lu, X. Wang, Study on Nanosized TiO<sub>2</sub>  
625 coated by silica in ultrasonic field, *Chin. J. Inorg Chem.* 21 (2005) 59-64.  
626 <https://doi.org/10.3321/j.issn:1001-4861.2005.01.010>
- 627 [34] X. Miao, H. Dai, J. Chen, J. Zhu, The enhanced method of hydroxyl radical  
628 generation in the heterogeneous UV-Fenton system with  $\alpha$ -FeOOH as catalyst,  
629 *Sep. Purif. Technol.* 200 (2018) 36-43.  
630 <https://doi.org/10.1016/j.seppur.2018.02.012>
- 631 [35] O. Gorduk, M. Gencten, S. Gorduk, M. Sahin, Y. Sahin, Electrochemical  
632 fabrication and supercapacitor performances of metallo  
633 phthalocyanine/functionalized-multiwalled carbon nanotube/polyaniline  
634 modified hybrid electrode materials, *J. Energy. Storage.* 33 (2020)  
635 102049-102061.  
636 <https://doi.org/10.1016/j.est.2020.102049>
- 637 [36] H. Liu, Y. Jiang, D. Liu, S. Li, X. Yang, Y. Ji, Y. Cui, Thermal effect on  
638 microstructure vibration of SiO<sub>2</sub> thin films, *Vib. Spectrosc.* 96 (2018) 101-105.  
639 <https://doi.org/10.1016/j.vibspec.2018.03.005>
- 640 [37] G.M. Rignanes, A. Pasquarello, J.C. Charlier, X. Gonze, R. Car, Nitrogen  
641 incorporation at Si(001)-SiO<sub>2</sub> interfaces: Relation between N 1s core-level shifts  
642 and microscopic structure, *Phys. Rev. Lett.* 79 (1997) 5174-5177.  
643 <https://doi.org/10.1103/PhysRevLett.79.5174>
- 644 [38] M. Xu, G. Pan, Y. Guo, S. Yang, Z. Fang, Surface modification and structure  
645 analysis of coated iron oxide yellow pigments to improve dispersion in organic  
646 solvents, *Surf. Interface Anal.* 53 (2021) 933-945.  
647 <https://doi.org/10.1002/sia.6995>

- 648 [39] D. Dai, X. Lan, L. Wu, Z. Wang, Designed fabrication of lightweight SiC/Si<sub>3</sub>N<sub>4</sub>  
649 aerogels for enhanced electromagnetic wave absorption and thermal insulation, J.  
650 Alloy. Compd. 901 (2022) 163651.  
651 <https://doi.org/10.1016/j.jallcom.2022.163651>
- 652 [40] A.N. Mansour, A. Brizzolara Robert, Characterization of the surface of  $\alpha$ -FeOOH  
653 powder by XPS, Surf. Sci. Spectra. 4 (1996) 357-357.  
654 <https://doi.org/10.1116/1.1247825>
- 655 [41] Y. Chae, J.H. Xin, T. Hua, Color prediction models for digital Jacquard woven  
656 fabrics, Color. Res. Appl. 41 (2016) 64-71.  
657 <https://doi.org/10.1002/col.21945>
- 658 [42] T. Thongkanluang, T. Kittiauchawal, P. Limsuwan, Preparation and  
659 characterization of Cr<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub>-Al<sub>2</sub>O<sub>3</sub>-V<sub>2</sub>O<sub>5</sub> green pigment, Thin. Solid Films.  
660 37 (2011) 543-548.  
661 <https://doi.org/10.1016/j.ceramint.2010.09.044>