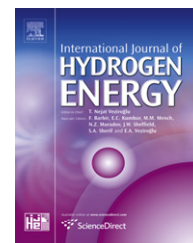


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# Static and dynamic hydrogen adsorption on Pt/AC and MOF-5

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## ABSTRACT

Hydrogen adsorption has been studied by static and dynamic methods on activated carbon (AC), platinum/activated carbon (Pt/AC), metal organic frameworks (MOF-5), and Pt/AC/MOF-5. The static method showed that all of adsorbents used in this study exhibited a Langmuir (type I) adsorption isotherm at 77 K and a linear function of hydrogen partial pressure at 298 K. The dynamic method produced breakthrough curves, indicating (i) slow rate of hydrogen diffusion in the densely packed activated carbon and Pt/AC beds and (ii) high rate of hydrogen diffusion in the loosely packed bed with large MOF-5 crystallites. Temperature variable adsorption resulted in the higher hydrogen uptake on Pt/AC than other adsorbents. The results suggested that temperature variable adsorption enhanced the hydrogen storage process by (i) initiating hydrogen dissociation at high temperature and (ii) facilitating spillover at low temperature on Pt/AC.

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## 1. Introduction

Hydrogen storage materials have been extensively studied by static adsorption methods. Static methods operate in either volumetric or gravimetric modes under equilibrium conditions [1,2]. The static method serves as an excellent approach to determine the number of adsorption sites, but provides little information on dynamic behaviors of hydrogen uptake/release processes. Practical hydrogen storage materials must possess not only high storage capacity but also rapid uptake/release rates. A potentially effective approach to investigate the process of hydrogen uptake/release is the dynamic approach which involves (i) varying either the inlet  $H_2$  concentration or adsorbent temperatures, and (ii) simultaneously monitoring a real-time response. This approach could provide valuable information of hydrogen storage and dynamic behavior of uptake/release.

We have employed a dynamic adsorption method to study hydrogen adsorption on Pt/AC (activated carbon) and MOFs

(metal-organic frameworks). Nanostructure materials such as activated carbon [3–5] and MOFs [6–9] with a high specific surface area (SSA) have been shown to exhibit high hydrogen storage capacity. Studies on various nanoscale carbon materials consisting of activated carbon [10–12] and carbon nanofibers [13–15] have indicated that hydrogen storage capacity is correlated with the pore volume and the SSA. Hydrogen storage capacity as high as 6 wt% have been reported for MOFs with a SSA of 3000 m<sup>2</sup>/g at 77 K [16]. Increasing temperature to 298 K reduces the hydrogen storage capacity to less than 1 wt % [17–20]. One approach to increasing hydrogen storage capacities at ambient conditions is to provide the sites for hydrogen adsorption by doping carbon materials with transition metals [21–24] and by mixing MOFs (metal-organic frameworks) with metal sites through building of the carbon bridge [25–27]. Molecule hydrogen could dissociate on the metal sites and then spillover (i.e., migration) from metal sites to the carbon or MOF surface.

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Nomenclature		m/e	mass/electron ratio
AC	activated carbon	MOFs or	MOF-5 metal organic frameworks
Ar	argon	MS	mass spectroscopy
BET	Brunauer-Emmet-Teller	Pt	platinum
C	concentrations	Pt/AC	platinum/activated carbon
D	deuterium	Pd/AC	pladium/activated carbon
DMF	N, N'-dimethylforamide	SSA	specific surface area
F(t)	$[C(t)-C_0]/[C_\infty - C_0]$ , normalized concentration	T	temperature, K
h	hour	t	time, min
H	hydrogen	TVA/D	temperature variable adsorption/desorption
He	helium	$\theta_H, \theta_D$	sites coverage
H-K	Horvath–Kawazoe method		

The dynamic adsorption method offers great advantages on adjusting adsorption parameters (i.e. concentration, temperature and pressure) and allows determination of kinetic characteristics of hydrogen adsorption. This paper reported results of a study on MOF-5, Pt/AC\_MOF-5, AC, and Pt/AC with three distinct dynamic adsorption methods: (i) a pulse of  $H_2$  into an inert gas stream, (ii) a gas switching between  $H_2$  and an inert gas stream, and (iii) a temperature variable adsorption (i.e. to adsorb during cooling from 298 to 77 K). Dynamic adsorption methods show hydrogen diffusion and adsorption behaviors by breakthrough curves during uptake. The temperature variable adsorption allowed hydrogen to overcome activation barrier of diffusion at 77 K and gave Pt/AC the highest  $H_2$  uptake.

## 2. Experimental

### 2.1. Synthesis of adsorbents

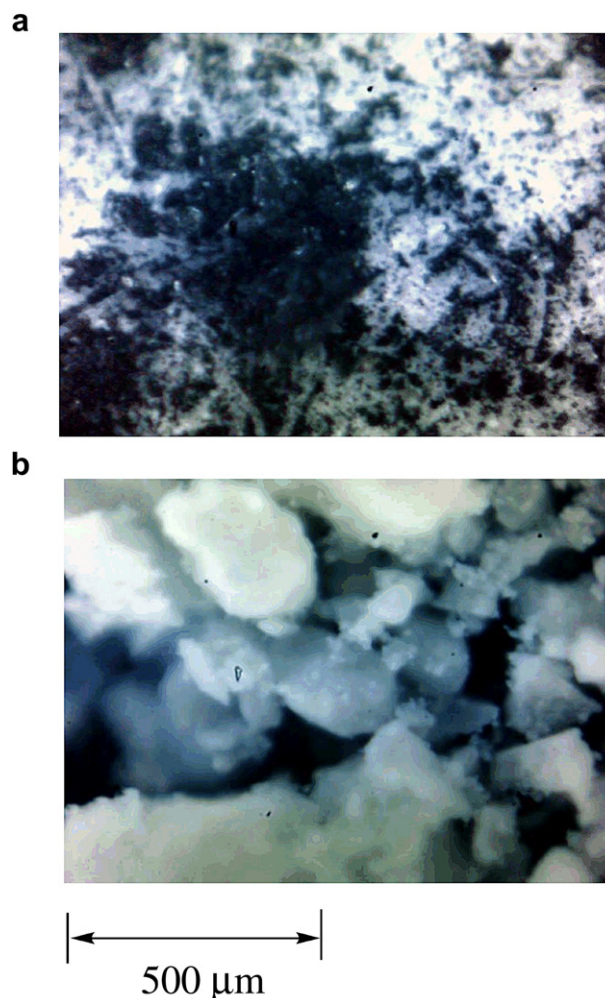
The metal organic frameworks (MOF-5) were synthesized by a fast mixing approach [28]. The synthesis consisted of (i) adding triethylamine (99.5%, Aldrich) into an N, N'-dimethylforamide (DMF, Aldrich) solution containing zinc nitrate hexahydrate (98%, Aldrich) and 1, 4-benzenecarboxylic acid (>99%, Aldrich), and (ii) vigorously stirring at room temperature for 4 h. The resulting precipitate from the synthesis was washed repeatedly with chloroform and filtered in vacuum at room temperature. The obtained precipitate was further degassed at room temperature for 6 h and then heated at 453 K under vacuum for 12 h. The final product was white crystallites as shown in Fig. 1 with a yield of 80 ( $\pm 5$ ) % based on zinc.

The 5 wt% Pt/AC was prepared by impregnating an aqueous solution of  $Pt(NH_3)_4(NO_3)_2$  (Aldrich) into activated carbon (pure graphite, Norit® SX Ultra CAT) under constantly stirring at room temperature for 30 min. The prepared Pt/AC was dried in air at 423 K and then reduced in flowing  $H_2$  at 773 K for 5 h. The final product form was fine black powder. The Pt/AC\_MOF-5 was obtained by physically mixing Pt/AC and MOF-5 in a weight ratio of 1: 9 for Pt/AC to MOF-5.

### 2.2. Characterization of adsorbents

The crystalline structures of Pt/AC and MOF-5 were determined by X-ray diffraction (XRD) with a Philips Analytical

X-Ray PW1710 diffractometer using Cu  $K\alpha$  radiation. The specific surface area, micropore volume, and pore width, of Pt/AC and MOF-5 were measured by an ASAP 2020 sorption meter (Micromeritics). The specific surface area was estimated by nitrogen adsorption isotherms at 77 K with the Brunauer-Emmet-Teller (BET) and Langmuir models. The micropore volume and pore width were obtained by the t-plot and Horvath–Kawazoe (H–K) method, respectively.



**Fig. 1** – Pictures of obtained (a) activated carbon and (b) MOF-5.

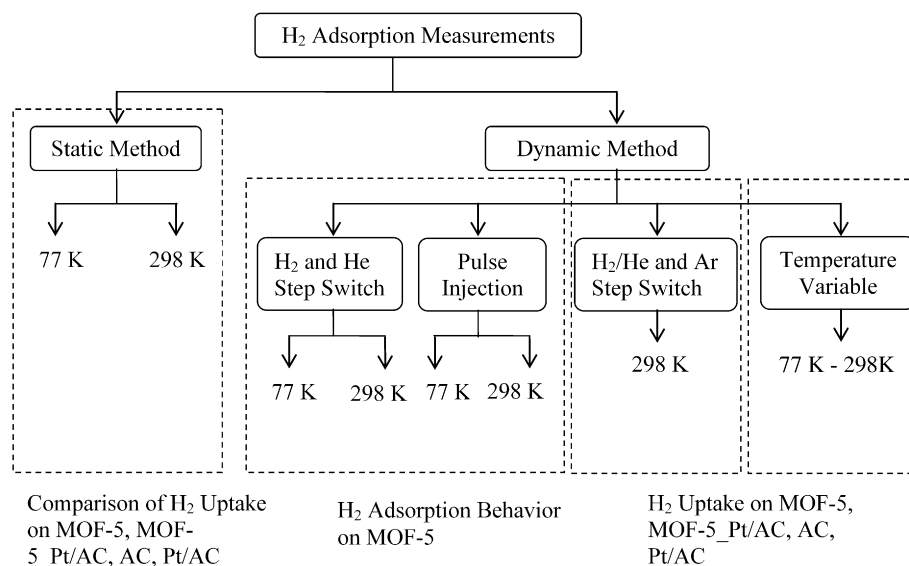


Fig. 2 – Design of adsorption experiments.

### 2.3. H<sub>2</sub> adsorption measurements

Fig. 2 illustrates the experimental approaches: static and dynamic adsorption methods. The static method was conducted on all adsorbents by an ASAP 2020 sorption meter at 77 and 298 K from 0 to 850 mmHg. Prior to measurements, 300(±10) mg of each adsorbent was degassed below 10  $\mu$ m Hg at 523 K for 6 h and Pt-containing adsorbents were further pre-treated with H<sub>2</sub> at 523 K. H<sub>2</sub> (UHP, Praxair) was purified through a molecular sieve (Grace Davison) before use.

The dynamic method was carried out in three modes: step switch, pulse injection, and temperature variable adsorption/desorption (TVA/D). The experimental apparatus, shown in Fig. 3, consists of (i) mass flow controllers (Brooks, Model 5850E) for adjusting the flow rates of H<sub>2</sub>, He and Ar (UHP, Praxair) streams, (ii) a quarter inch I.D. U shape tubular column filled with 100 (±5) mg of adsorbent particles which were packed with quartz wool at both ends, and (iii) a quadrupole mass spectrometer (MS; Pfeiffer vacuum, Omnistar GSD 301) with a differentially pumped gas inlet. The inlet of MS allowed a portion of the effluents at 1 atm to enter an

ionization chamber at approximately 0.7 mbar. The mass/electron ratios, i.e.  $m/e$ , were selected for H<sub>2</sub> (2), HD (3), D<sub>2</sub> (4), He (4), and Ar (40). The calibration factor for each gaseous species, i.e. the ratio of the area under the MS profile to the amount of species injected, was determined by pulse injections of known amount of species.

Step switch was achieved by a 4-port valve which gives a step change in gaseous concentrations while maintaining the total flow rate at steady state conditions. Pulse injections were obtained by a 6-port valve which injects 5 cm<sup>3</sup> of H<sub>2</sub> into the carrier gas stream, entering to the adsorption column. The temperature variable adsorption/desorption (TVA/D) was performed by exposing the adsorbent to H<sub>2</sub> during cooling from 298 to 77 K and then warming from 77 to 298 K.

Isotope exchange experiments used for verification of H<sub>2</sub> dissociation were performed by (i) switching between H<sub>2</sub>/D<sub>2</sub> (50/50) and Ar flow and (ii) a pulse injection of D<sub>2</sub> into H<sub>2</sub> gas flow through the adsorption column at 298 K. Deuterium (99.6% D<sub>2</sub> with 0.04% HD,) was purchased from Cambridge Isotope laboratory.

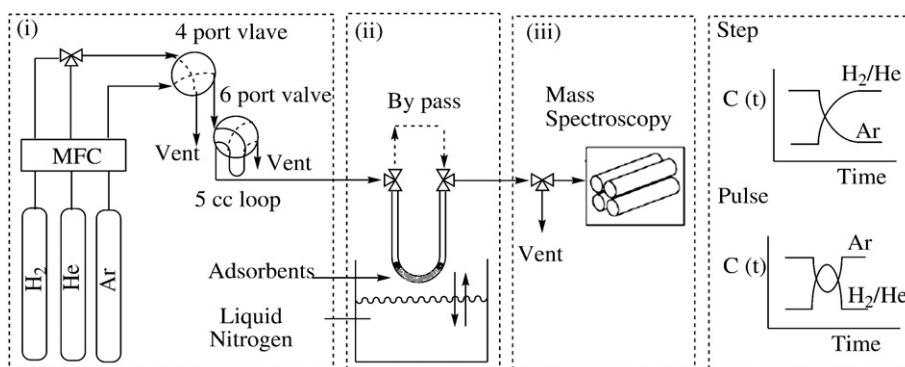


Fig. 3 – The experimental apparatus for dynamic hydrogen adsorption.

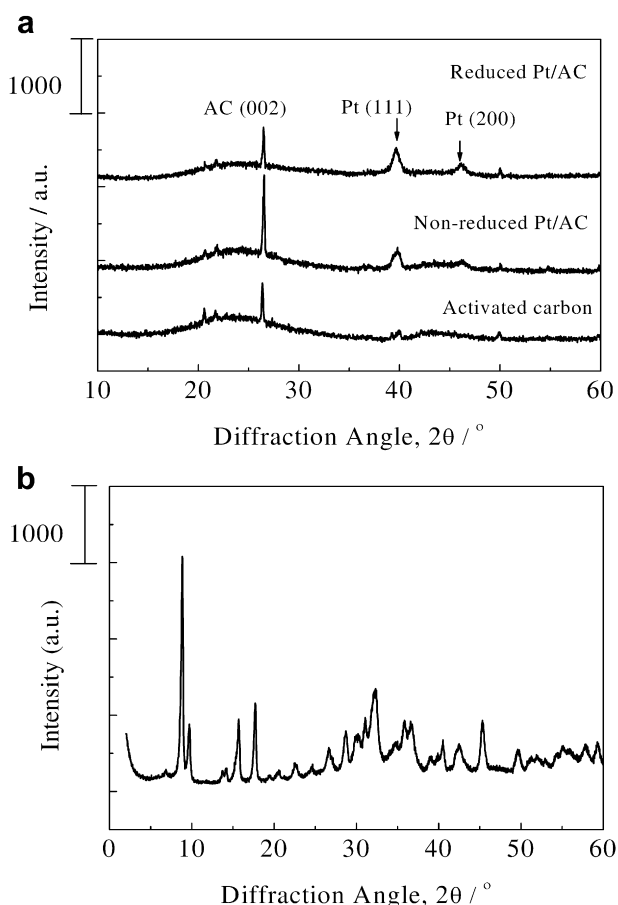


Fig. 4 – X-ray diffraction patterns of (a) activated carbon, non-reduced and reduced Pt/AC and (b) MOF-5.

### 3. Results and discussion

#### 3.1. Characterization

Fig. 4 shows the XRD pattern for AC, Pt/AC and MOF-5. A diffraction peak at  $26.4^\circ$  on both AC and Pt/AC was ascribed to graphite (002) [29]. The diffraction peaks at  $39.6^\circ$  and  $46.2^\circ$  on Pt/AC were assigned to Pt (111) and Pt (200), respectively [30–32]. The XRD pattern of MOF-5 in Fig. 4(b) gave a lattice parameter of  $25.8 \text{ \AA}$ , which was consistent with experimental [20,33] and simulated results [34]. The surface

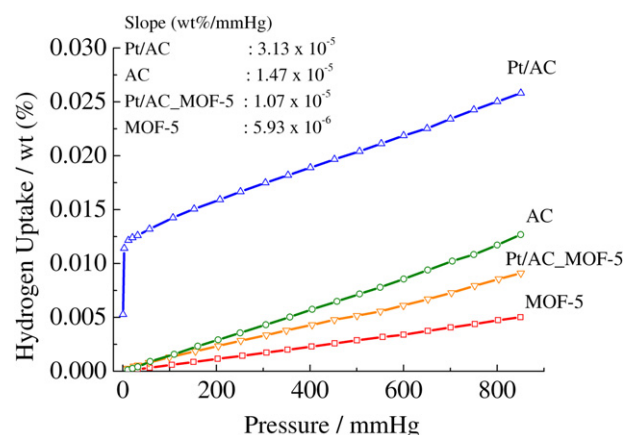


Fig. 5 – Hydrogen adsorption isotherms of carbon adsorbents and MOF-5 at 298 K.

characterizations on these adsorbents are summarized in Table 1. The BET surface area, micropore and pore volume of MOF-5 agreed with those reported in previous literature results [20,28].

#### 3.2. $H_2$ storage and adsorption isotherm

Fig. 5 shows hydrogen adsorption isotherms for all adsorbents at 298 K. All hydrogen adsorption isotherms except Pt/AC exhibited a linear behavior, obeying Henry's law [10,35]. The slope of adsorption isotherms corresponds to an adsorption equilibrium constant, which reflects the strength of interaction between adsorbate and adsorbent. The hydrogen adsorption isotherm on Pt/AC at low pressures indicated the occurrence of chemisorption, which was also observed on other Pd/AC adsorbents and AC-supported metals [36,37]. Hydrogen uptake on Pt/AC at the turning point of slope in the adsorption isotherm was  $1.14 \times 10^{-2} \text{ wt\%}$  (corresponding to a ratio  $3.81$  of  $H_{\text{ads}}/\text{Pt}$ ), which was higher than  $3.0 \times 10^{-3} \text{ wt\%}$  ( $H_{\text{ads}}/\text{Pt} = 1$ ). The high ratio of  $H_{\text{ads}}/\text{Pt}$  on Pt/AC suggested spillover of adsorbed hydrogen on Pt to the carbon surface [38]. Hydrogen uptake on all adsorbents increased in the order: MOF-5 < Pt/AC\_MOF-5 < AC < Pt/AC. A relatively high Henry's constant (i.e. the slope of isotherm) for activated carbon was believed to be responsible for substantial hydrogen storage (i.e.,  $6.5 \text{ wt\%}$ ) on carbon nanofibers [13,21]. In contrast, a low Henry's constant for MOF-5 corresponded to low hydrogen

Table 1 – Surface characterization on carbon adsorbents and MOF-5.

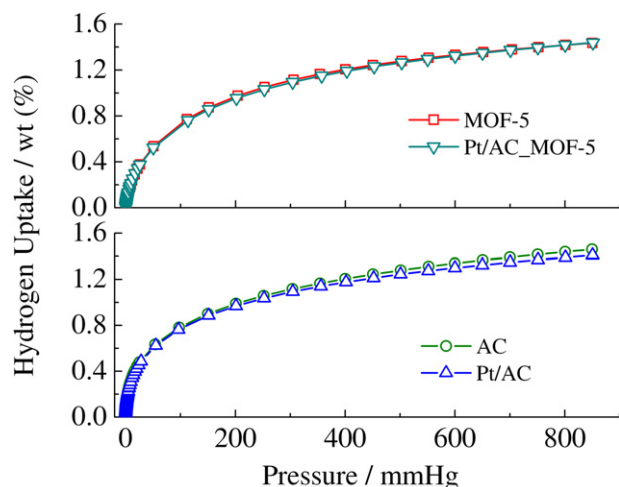
Sample	BET SA ( $\text{m}^2/\text{g}$ ) <sup>a</sup>	Langmuir SA ( $\text{m}^2/\text{g}$ ) <sup>a</sup>	Micropore volume ( $\text{cm}^3/\text{g}$ ) <sup>b</sup>	Pore volume ( $\text{cm}^3/\text{g}$ ) <sup>c</sup>	Medium Pore width (nm) <sup>c</sup>
Activated Carbon (AC)	1050	1240	0.26	0.94	3.20
5 wt% Pt/AC	990	1110	0.24	0.87	3.18
MOF-5	670	860	0.25	0.30	0.79
5 wt% Pt/AC_MOF-5	740	840	0.16	0.31	0.80

a Error =  $\pm 5\%$ .

b From t-plot analysis.

c From H-K analysis.





**Fig. 6 – Hydrogen adsorption isotherms of carbon adsorbents and MOF-5 at 77 K.**

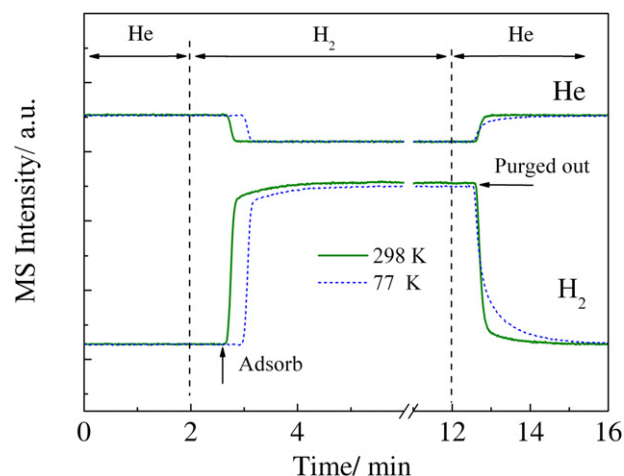
uptake, giving less than 0.2 wt% at pressure up to 67 bar [20] (0.3% as extrapolated in our studies). The low Henry's constant reflected that the interaction between  $H_2$  and MOF-5 is in the form of van der Waals force. The monolayer adsorption of carbon adsorbents with a surface area  $1000 \text{ m}^2/\text{g}$  will give a hydrogen uptake of 2.28 wt%, according to a theoretical value  $2.28 \times 10^{-3} \text{ (mass\%/m}^2\text{g)}$  [39]. Therefore, the monolayer adsorption of hydrogen on all adsorbents was not achieved.

Fig. 6 shows hydrogen adsorption isotherms on all adsorbents at 77 K. All isotherms displayed a type I (Langmuir) isotherm characteristics of microporous materials. The isotherm of MOF-5 resembled those previously reported in literature [20,25]. The saturation point of adsorption isotherms was not observed in the range of pressure investigated, implying that hydrogen uptake could be further increased by increasing  $H_2$  pressure. The saturation point of hydrogen uptake on MOF-5 at 77 K has been reported to be 5.1 wt% [40] at the pressure above 50 bar. All hydrogen adsorption isotherms on different adsorbents, regardless of the presence or absence of Pt, at 77 K, showed the same trend with significant overlapping, suggesting a characteristic of physisorption, which exhibits low activation energy and a nonspecific interaction with adsorbents. Hydrogen uptake during physisorption can be correlated to the surface area accessible to  $H_2$  molecules [10,20] and decreases rapidly as temperature increases.

**Table 2 – The hydrogen storage capacity in the static method.**

Adsorbent	Static adsorption		
	77 K	77 K (reported)	298 K (wt%)
MOF-5	1.399	1.3–1.5 <sup>20,24</sup>	5.47E-03
Pt/AC_MOF-5	1.395		9.20E-03
AC	1.415	1.4–1.5 <sup>12,14</sup>	1.27E-02
Pt/AC	1.369		2.58E-02

Note: error of storage capacity =  $\pm 5\%$ .

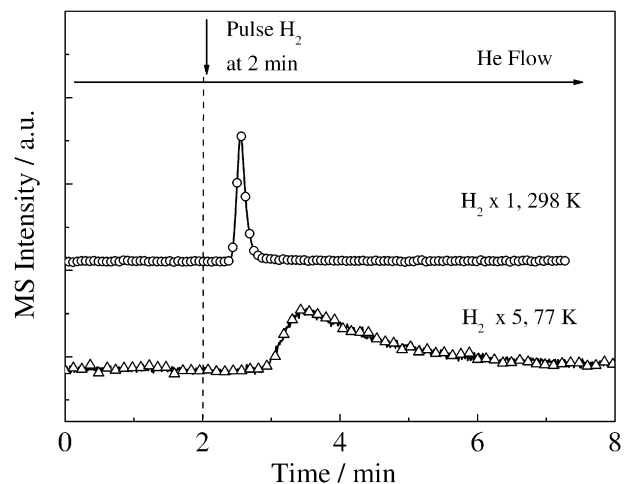


**Fig. 7 – MS profiles of step switching between He and  $H_2$  gas flow through MOF-5 at 77 and 298 K.**

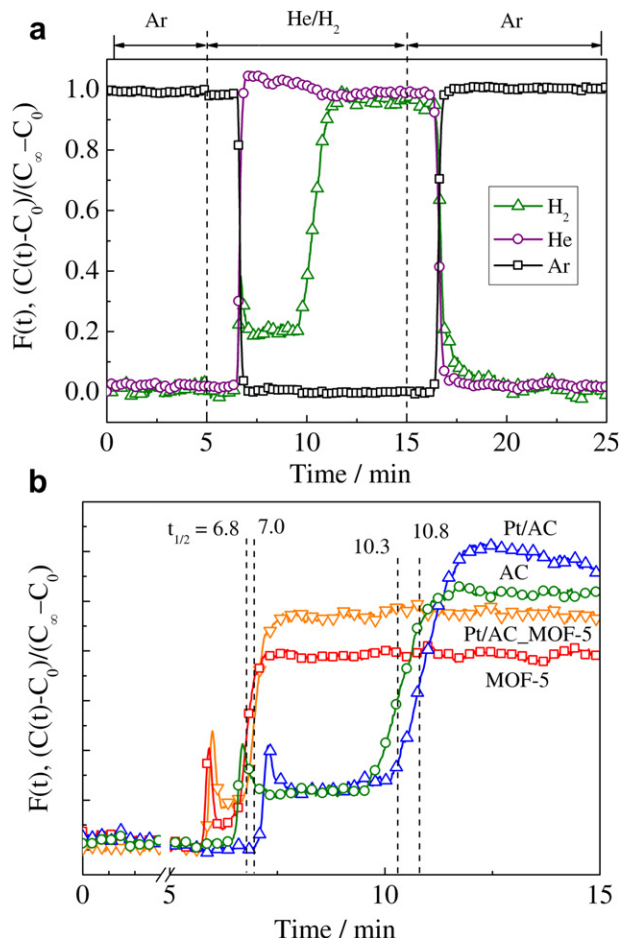
Table 2 lists the hydrogen storage capability on all adsorbents in this study and compares with those reported in the literature. The hydrogen storage capacity normalized by the surface area at 77 K was comparable to literature values [25,41,42]. Although Pt is able to enhance hydrogen uptake on activated carbon at 298 K, such an enhancement was not observed at 77 K. This could be attributed to the absence of  $H_2$  dissociation and subsequent spillover since dissociation of  $H_2$  on the Pt surface has a high activation energy. It has been shown that ultrahigh vacuum treatment at 1073 K is needed to generate Pt sites for dissociative adsorption of  $H_2$  [43].

### 3.3. The dynamic method

Fig. 7 shows the MS profiles of step switching from a stream of He to  $H_2$  flowing into the adsorption column filled with MOF-5 at 77 and 298 K. The nearly perfect step change in the He profile indicated the absence of adsorption/desorption. In contrast, the  $H_2$  profile rose rapidly, followed by a gradual



**Fig. 8 – MS profiles during  $H_2$  pulses through MOF-5 at 77 and 298 K.**



**Fig. 9** –  $H_2$ , He and Ar,  $F(t)$  profiles during (a) switching between 5%  $H_2$ /He and Ar and (b)  $H_2$  adsorption over MOF-5 ( $\square$ ), Pt/AC\_MOF-5 ( $\nabla$ ), AC ( $\circ$ ), and Pt/AC ( $\Delta$ ) at 298 K.

increase to a final value. A similar  $H_2$  profile with an opposite direction was also observed upon switching gas flow from  $H_2$  to He. Decreasing temperature from 298 to 77 K prolonged trailing of the  $H_2$  profile. The trailing profile was also observed in the pulse injection of  $H_2$  into He at 77 K in Fig. 8. The significant trailing indicated a low rate of diffusion and a high hydrogen uptake.

To distinguish between hydrogen uptake and relative rates of hydrogen adsorption on various adsorbents, the step switch was conducted by switching a stream of  $H_2$ /He (5%  $H_2$ ; He is a tracer) to Ar. Fig. 9(a) shows  $F(t)$  profiles of  $H_2$ , He, and Ar obtained during step switch at 298 K.  $F(t)$ , a normalized concentration profile, is obtained by [44,45],

$$F(t) = \frac{C(t) - C_0}{C_\infty - C_0}$$

where  $C(t)$  is the concentration of  $H_2$ ;  $C_0$  and  $C_\infty$  are  $H_2$  concentration at  $t = 0$  and steady state, respectively. All of adsorbents showed a sharp increase followed by a gradual increase in  $F(t)$  profiles. The first sharp increase, denoted as the first breakthrough curve, can be attributed to the hydrogen flow which passed through a loosely packed region of the adsorbent bed. The second breakthrough curve can be attributed to the hydrogen flow passing through a tightly

**Table 3** – The hydrogen uptake and release in the dynamic method.

Adsorbent	Adsorption at 298 K <sup>a</sup>		TVA/D <sup>b</sup>	
	Uptake	$t_{1/2}$	Uptake at 77 K	Release at 298 K (wt%)
MOF	0.071	6.8	0.771	0.471
Pt/AC_MOF	0.103	7.0	0.796	0.489
AC	0.117	10.3	0.781	0.789
Pt/AC	0.189	10.8	1.134	1.041

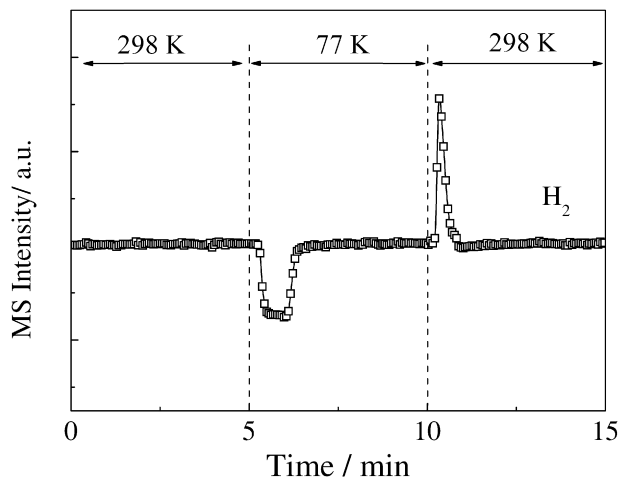
<sup>a</sup> Data of Fig. 9.

<sup>b</sup> Data of Fig. 11; error:  $\pm 5\%$ .

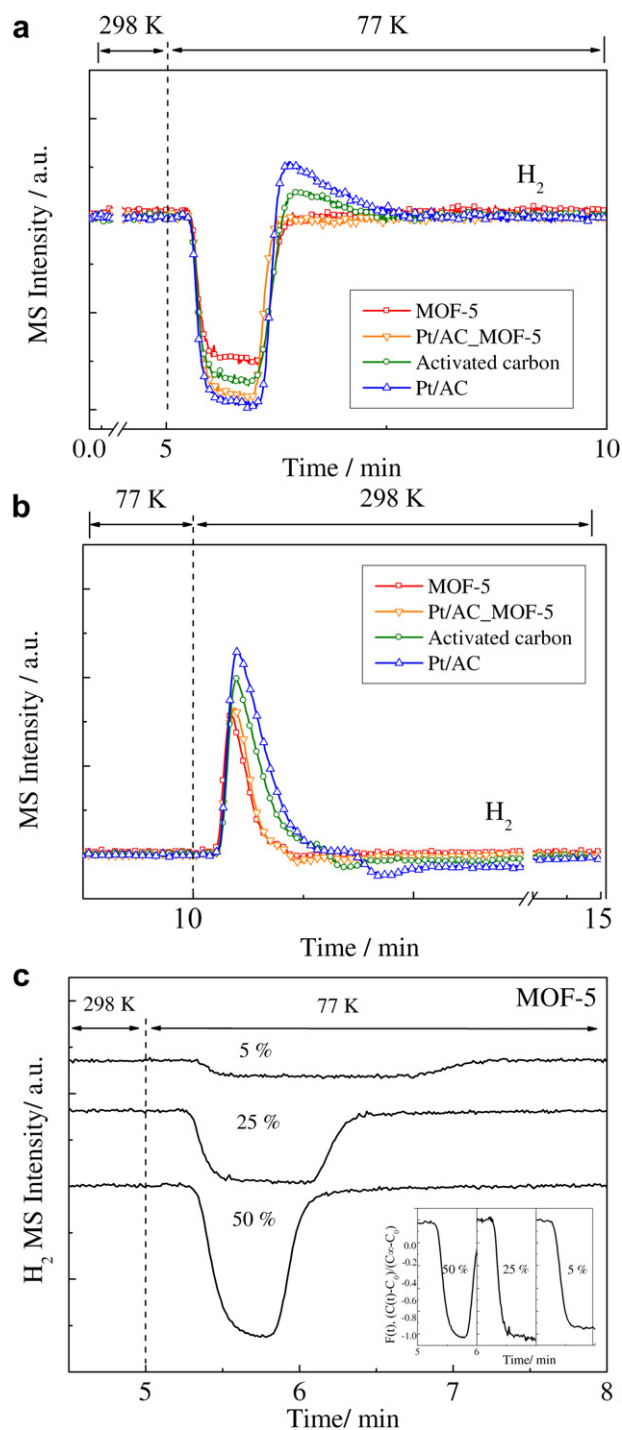
packed region. The slope of the breakthrough curve characterizes the rate of diffusion. Higher diffusion rate would give a breakthrough curve with a sharper slope and a shorter half-time of diffusion (i.e.  $t_{1/2}$ ). The hydrogen adsorption capacities determined by the area between the He and  $H_2$  breakthrough curves are listed in Table 3. The hydrogen uptake increased in the order: MOF < Pt/AC\_MOF-5 < AC < Pt/AC. The hydrogen uptake determined from the step switch method followed the same trend as those determined by the static method at 298 K in Fig. 5 and Table 3, but the former method gave higher hydrogen storage capacity than the latter due to higher  $H_2$  partial pressures.

Fig. 10 illustrates temperature variable adsorption and desorption of  $H_2$  (25%) in He on MOF-5. The adsorbent was exposed to  $H_2$ /He flow at 298 K. At  $t = 5$  min, the adsorbent column was lowered down to 77 K in liquid nitrogen; at  $t = 10$  min the adsorbent column was lifted back to 298 K. Decreasing temperature from 298 to 77 K caused a rapid drop in the  $H_2$  profile due to increasing hydrogen adsorption; raising temperature back to 298 K caused a sharp rise due to increasing hydrogen desorption. Comparison of  $H_2$  profiles on various adsorbents in Fig. 11 shows that Pt/AC exhibited the highest hydrogen uptake and release.

The  $H_2$  uptake and release obtained in TVA/D are presented on Table 3. The higher uptake than release was a result of the transient nature of the experiment where hydrogen adsorbed

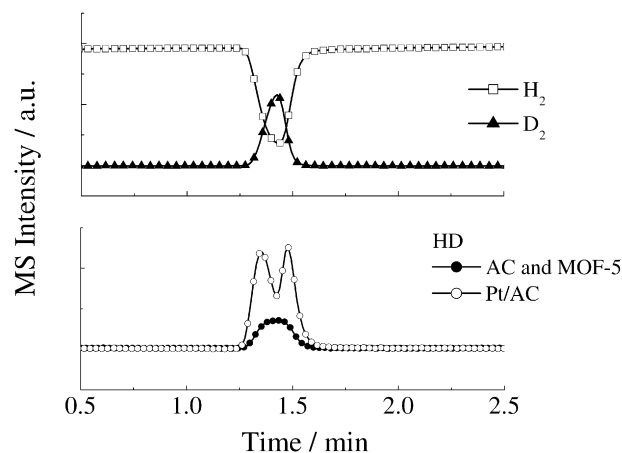


**Fig. 10** –  $H_2$  MS profiles during temperature variable adsorption and desorption on MOF-5 in 25%  $H_2$ /He.



**Fig. 11** – H<sub>2</sub> MS profiles during (a) temperature variable adsorption of H<sub>2</sub> at 77 K, (b) desorption at 298 K on all adsorbents in a 25% H<sub>2</sub>/He gas stream, and (c) different conc. of H<sub>2</sub> adsorption on MOF-5.

during uptake was gradually released at a rate which was insufficient to give a rise in H<sub>2</sub> profiles. The adsorption kinetics was characterized by the slope of MS adsorption profile with different concentrations of H<sub>2</sub>, e.g. MOF-5 in Fig. 11(c). Higher H<sub>2</sub> concentrations gave a steeper MS profile (inset of Fig. 11(c)), indicating higher rates of adsorption and diffusion. Comparison of hydrogen uptake obtained by dynamic methods in



**Fig. 12** – Isotope exchange of deuterium during pulse injection of D<sub>2</sub> in H<sub>2</sub> flow through Pt/AC, AC and MOF-5 at 298 K.

Table 3 shows that the TVA/D approach resulted in higher hydrogen storage on Pt/AC, suggesting that thermal energy possessed by H<sub>2</sub> at 298 K may allow it to overcome the activation barrier for H<sub>2</sub> adsorption during cooling of the adsorbent bed to 77 K.

The higher H<sub>2</sub> uptake of Pt/AC can be attributed to the dissociative adsorption of hydrogen followed by spillover of adsorbed hydrogen from Pt to AC sites. The dissociative adsorption of hydrogen on Pt/AC can be further evidenced by the formation of two humps of HD during the pulse injection of D<sub>2</sub> into H<sub>2</sub> over Rh/SiO<sub>2</sub> catalysts [46]. The observed equal size of two humps in HD response is due to the rapid dissociation of H<sub>2</sub> and D<sub>2</sub> followed by the combination of adsorbed H and D. The rapid dissociative hydrogen and deuterium adsorption allowed the coverage of adsorbed H and D on adsorbents closely follow the concentration of H<sub>2</sub> and D<sub>2</sub>. The peak of the hump would occur at nearly equal coverage of H and D ( $\theta_H$  and  $\theta_D$ ), assuming that  $r_{HD} = k \cdot \theta_H \cdot \theta_D$ . As the coverage of H and D deviates from equal coverage,  $r_{HD}$  decreases.

Hydrogen dissociation on Pt sites and subsequent spillover from Pt to carbon matrix has been reported [47]. Crystallite face of Pt (111) has the lowest surface energy [48,49] and adsorbs hydrogen with heat of adsorption energy about 237 (kJmol<sup>-1</sup>) [38]. These hydrogen have been shown to be responsible for H<sub>2</sub> spillover and oxidation reactions [50]. At a constant temperature, hydrogen spillover can be a reversible process which allows adsorbed hydrogen to reversely migrate from carbon surface back to Pt surface for desorption [51]. The reversible nature suggested that the spillover process has lower activation energy than H<sub>2</sub> chemisorption. The activation energy of H<sub>2</sub> spillover has been determined to be less than 10 kJmol<sup>-1</sup> [52]. The reversed spillover process may contribute to overshoot of H<sub>2</sub> MS profiles during hydrogen desorption on Pt/AC in Fig. 11(a). The spillover and reversed spillover process did not occur on the physical mixture of Pt/AC and MOF, as evidenced by its low hydrogen storage capacity. This result suggested that adsorbed hydrogen on the carbon surface was not able to transfer via

a physical contact between carbon and MOF surfaces, confirming the results of a number of previous studies [37].

## 4. Conclusions

The hydrogen adsorption isotherm on all adsorbents showed type I (Langmuir) isotherm characteristics of microporous materials at 77 K, giving a hydrogen storage capacity in the range of 1.3–1.4 wt%. The results suggested that hydrogen adsorption on these materials was in the form of physisorption. At 298 K, hydrogen adsorption capacities on all adsorbents gave a linear function of pressure with the capacity increasing in the order: MOF-5 < Pt/AC\_MOF-5 < AC < Pt/AC. The dynamic method at 298 K showed that small Pt/AC particles gave a close packed bed, slowing down the diffusion of H<sub>2</sub> to the adsorption sites and exhibiting a breakthrough curve with a smaller slope than those resulted from the loosely bed packed by large MOF particles. Pt/AC exhibited a significantly higher hydrogen storage capacity during temperature variable adsorption than dynamic step and static adsorption at 298 K, suggesting that hydrogen adsorption during TVA/D involves two steps: (i) the H<sub>2</sub> dissociative step at high temperature and (ii) the subsequent spillover step. Temperature variable adsorption can serve as an approach to enhance the adsorption process which involves a number of steps with different activation energies.

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