Equations for a growing (and thickening) cambial cell with a multilayered wall including the effect of cellulose microfibrils reorientation

Cyril Bozonnet, Tancrède Alméras, Joseph Gril, Félix Hartmann

May 15, 2024

1 Main objective

Explore the effect of the wall heterogeneity on the growth of a cambial cell.

2 Geometry and kinematics

2.1 At the cell scale

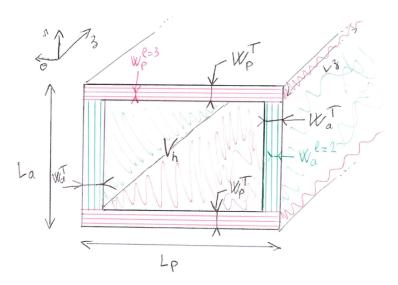


Figure 1: Geometry and notations. The reference frame at the top left corner shows the link between the present geometry and the one of an actual cambial cell whose growth would be mainly along r.

The geometry of the cambial cell is displayed in figure 1. We make the following additional assumptions:

- \bullet Growth occurs only along r
- Wall layer deposition occurs only along the anticlinal walls
- Wall synthesis is regulated so that the total anticlinal wall thickness is constant

Under the previous hypothesis, the elongation rate in the radial (anticlinal) direction is

$$\bar{\varepsilon}_a = \frac{1}{V_h} \frac{\mathrm{d}V_h}{\mathrm{d}t} = \frac{Q}{V_h} = \frac{A\kappa_h}{V_h} \left(\Psi_{src} - P + \Pi \right), \tag{1}$$

where V_h is the lumen volume, Q the sum of water fluxes entering or leaving the cell, A the exchange area for the water fluxes, κ_h the hydraulic conductivity, Ψ_{src} , P, and Π are the total external, the cell pressure and cell osmotic potentials, respectively. The cell pressure is computed from the mean anticlinal stresses, $\overline{\sigma_a}$, as

$$P = \frac{2\overline{\sigma_a}W_a^T}{L_p}. (2)$$

2.2 At the wall scale

The wall structure is formed by multiple layers spreaded onto each other. Each layer is made up of cellulose microfibrils (MF), that are inclined at an angle ϕ with respect to the longitudinal direction, and surrounded by a gel composed of different polymers. These MF are deposited at a given angle, ϕ_0^i . The MF angle in a wall layer ϕ^i evolves with the total wall layer deformation, ε_T^i according to

$$\tan(\phi^i) = \tan(\phi_0^i) \exp(\varepsilon_T^i). \tag{3}$$

The total layer deformation is computed as $\varepsilon_T^i = (L_a - L_0^i)/L_0^i$, where L_0^i is the length of the layer when it was deposited and L_a is the actual anticlinal cell length (obtained by integrating Eq. (1)).

3 Mechanics of a growing heterogeneous cell wall

3.1 General procedure

In order to describre properly the behaviour of this heterogeneous medium we proceed in different steps:

- 1. From the cell boundary conditions and the kinematics, we know the deformation rate tensor in the reference frame (R,Z) for each layer.
- 2. Through a first rotation step (of angle $\pi/2 \phi^i$), one can express the previous tensor in the MF frame of each layer $(1,2)^i$.
- 3. Appplying a consitutive equation, one can then obtain the changes in the stress fields in each layer in the MF frame $(\dot{\sigma_1}, \dot{\sigma_2}, \dot{\tau_{12}})^i$. The consitutive equation will take into account the heterogenous nature of the wall layer.
- 4. Through a second rotation step (of angle $\phi^i \pi/2$), one can express the changes in the stress fields from the rotated frame to the reference frame, for each wall layer, and integrate them to get the stress fields.
- 5. Finally, the mean anticlinal wall stress can be computed, frow which one can compute the new pressure, that will then be used to compute the new elongation rate.

This procedure is not restricted to mono-dimensional growth and could be applied in case of intrusive growth as well.

3.2 Deformation rate tensor

Assuming that the cell deformation is blocked in the longitudinal direction, that there is no distortion in the (R,Z) frame, the cell deformation rate tensor can be written, in a vector form,

$$\bar{\bar{\varepsilon}}_{R,Z} = \begin{pmatrix} \dot{\varepsilon}_R \\ \dot{\varepsilon}_Z \\ \dot{\gamma}_{RZ2}/2 \end{pmatrix} = \begin{pmatrix} \dot{\varepsilon}_a \\ 0 \\ 0 \end{pmatrix}.$$
(4)

This expression is true for all wall layers.

3.3 Rotation steps

The elongation rate tensor in a wall layer can be expressed in the rotated frame (rotation of angle θ) through the following operation

$$\begin{pmatrix} \dot{\varepsilon}_1 \\ \dot{\varepsilon}_2 \\ \dot{\gamma}_{12}/2 \end{pmatrix} = \begin{pmatrix} \cos^2(\theta) & \sin^2(\theta) & 2\sin(\theta)\cos(\theta) \\ \sin^2(\theta) & \cos^2(\theta) & -2\sin(\theta)\cos(\theta) \\ -\sin(\theta)\cos(\theta) & \sin(\theta)\cos(\theta) & \cos^2(\theta) - \sin^2(\theta) \end{pmatrix} \begin{pmatrix} \dot{\varepsilon}_a \\ 0 \\ 0 \end{pmatrix}. \tag{5}$$

Replacing θ by $\pi/2 - \phi^i$ leads to

$$\begin{pmatrix}
\dot{\varepsilon}_1 \\
\dot{\varepsilon}_2 \\
\dot{\gamma}_{12}/2
\end{pmatrix}^i = \begin{pmatrix}
\sin^2(\phi^i) & \cos^2(\phi^i) & 2\cos(\phi^i)\sin(\phi^i) \\
\cos^2(\phi^i) & \sin^2(\phi^i) & -2\cos(\phi^i)\sin(\phi^i) \\
-\cos(\phi^i)\sin(\phi^i) & \cos(\phi^i)\sin(\phi^i) & \sin^2(\phi^i) - \cos^2(\phi^i)
\end{pmatrix} \begin{pmatrix}
\dot{\varepsilon}_a \\
0 \\
0
\end{pmatrix},$$
(6)

which must be computed for every wall layer.

The stress tensor components in the reference frame can be computed from the stress tensor components in the MF frame using a rotation of angle $-\theta = -(\pi/2 - \phi^i)$,

$$\begin{vmatrix} \dot{\sigma}_a \\ \dot{\sigma}_Z \\ \dot{\tau}_{RZ} \end{vmatrix}^i = \begin{pmatrix} \sin^2(\phi^i) & \cos^2(\phi^i) & -2\cos(\phi^i)\sin(\phi^i) \\ \cos^2(\phi^i) & \sin^2(\phi^i) & 2\cos(\phi^i)\sin(\phi^i) \\ \cos(\phi^i)\sin(\phi^i) & -\cos(\phi^i)\sin(\phi^i) & \sin^2(\phi^i) - \cos^2(\phi^i) \end{pmatrix} \begin{pmatrix} \dot{\sigma}_1 \\ \dot{\sigma}_2 \\ \dot{\tau}_{12} \end{pmatrix}^i . \tag{7}$$

3.4 Constitutive equation

3.4.1 General form

We assume that the mechanical behaviour of the ensemble fibre + gel within each wall layer is similar to a Bingham fluid and that we are in plane stress conditions, i.e.,

$$\left[\begin{pmatrix} \dot{\sigma}_1 \\ \dot{\sigma}_2 \\ \dot{\tau}_{12} \end{pmatrix}^i = \overline{\overline{Q}} \begin{pmatrix} \dot{\varepsilon}_1 \\ \dot{\varepsilon}_2 \\ \dot{\gamma}_{12} \end{pmatrix}^i - \frac{1}{\tau_{visc}} \left[\begin{pmatrix} \sigma_1 \\ \sigma_2 \\ \tau_{12} \end{pmatrix}^i - \begin{pmatrix} \beta_1 \\ \beta_2 \\ \beta_{12} \end{pmatrix}^i \right]^+, \tag{8}$$

where $\overline{\overline{Q}}$ is the rigidity tensor in plane stress conditions (it is the same for all layers), τ_{visc} is the characteristic viscous time (supposedly the same for all stress components), β_1 , β_2 and β_{12} are the different yield thresholds, and the $(x)^+$ notation is equivalent to $\max(x,0)$. The rigidity tensor under plane stress hypothesis is expressed as

$$\overline{\overline{Q}} = \begin{pmatrix} Q_{11} & Q_{12} & 0 \\ Q_{12} & Q_{22} & 0 \\ 0 & 0 & Q_{66} \end{pmatrix} = \begin{pmatrix} \frac{E_1}{1 - \nu_{12}\nu_{21}} & \frac{\nu_{21}E_1}{1 - \nu_{12}\nu_{21}} & 0 \\ \frac{\nu_{21}E_1}{1 - \nu_{12}\nu_{21}} & \frac{E_2}{1 - \nu_{12}\nu_{21}} & 0 \\ 0 & 0 & G_{12} \end{pmatrix},$$
(9)

where E_1 , E_2 , ν_{12} , ν_{21} and G_{12} , are the Young's modulus, Poisson coefficients, and the shear modulus in the MF frame, respectively.

3.4.2 Elastic parameters

To compute the elastic coefficients of the fibre + gel (matrix) ensemble, we use the Halpin-Tsai equations, that are widely used to predict the elastic properties of polymers reinforced by short fibers (Halpin and Kardos, 1976; Abrate, 1986). One can thus compute the elastic properties using the following expressions:

$$E_{1} = E_{m} \frac{1 + \xi \frac{1}{d} \eta_{L} V_{f}}{1 - \eta_{L} V_{f}},$$

$$E_{2} = E_{m} \frac{1 + \xi \eta_{T} V_{f}}{1 - \eta_{T} V_{f}},$$

$$G_{12} = G_{m} \frac{1 + \xi \eta_{G} V_{f}}{1 - \eta_{G} V_{f}},$$

$$\nu_{12} = \nu_{m} \frac{1 + \xi \eta_{\nu} V_{f}}{1 - \eta_{\nu} V_{f}},$$

$$\nu_{21} = \nu_{12} \frac{E_{2}}{E_{1}},$$

$$(10)$$
where $\eta_{L} = \frac{(E_{f}/E_{m} - 1)}{E_{f}/E_{m} + \xi l/d},$

$$\eta_{T} = \frac{(E_{f}/E_{m} - 1)}{E_{f}/E_{m} + \xi},$$

$$\eta_{G} = \frac{(G_{f}/G_{m} - 1)}{G_{f}/G_{m} + \xi},$$
and $\eta_{\nu} = \frac{(\nu_{f}/\nu_{m} - 1)}{\nu_{f}/\nu_{m} + \xi}.$

The different parameters, their significance and indicative values are reported in table 1.

Variable name	Description	Unit	Value or expression
Material and geometrical properties			
E_m	Matrix Young's modulus	GPa	0.01
E_f	MF Young's modulus	GPa	198 (Song et al., 2021)
$ u_m$	Matrix Poisson coefficient	-	0.25
$ u_f$	MF Poisson coefficent	-	0.22 (Song et al., 2021)
G_m	Matrix shear modulus	GPa	$E_m/(2(1+\nu_m))$
G_f	MF shear modulus	GPa	6.06 (Song et al., 2021)
V_f°	MF volume fraction	-	0.1
$V_f \ \xi$	Reinforcement parameter	-	1.5 (for random organization)
l/d	MF aspect ratio	-	10^{2}
Elastic parameters computed from Halpin-Tsai equations using above values			
E_1	Axial Young's modulus	MPa	176.3
E_2	Transverse Young's modulus	MPa	12.77
$ u_{12}$	Poisson coefficient	-	0.2911
ν_{21}	Poisson coefficient	-	0.02
G_{12}	Shear modulus	MPa	4.9

Table 1: Elastic parameters

3.4.3 Plastic thresholds

Finally, one must compute β_1^i , β_2^i and β_{12}^i , the plastic thresholds that appear in Eq. (8). To determine whether or not the stresses are above the yield threshold, one must use in the MF frame a criterion adapted to the anisotropic nature of the wall material. For this, we use the Tsai-Hill criterion, that has been widely used for anisotropic composite materials (Tsai, 1965; Fuchs et al., 2006; Mortazavian and Fatemi, 2015). This criterion states that when the following inequality is violated, plasticity starts:

$$\left(\frac{\sigma_1^i}{Y_1}\right)^2 - \frac{\sigma_1^i \sigma_2^i}{Y_1^2} + \left(\frac{\sigma_2^i}{Y_2}\right)^2 + \left(\frac{\tau_{12}^i}{Y_{12}}\right)^2 < 1.$$
(11)

In the previous equation, Y_1 and Y_2 are the tensile or compressive yield stress in the 1 and 2 directions, and Y_{12} is the shear yield stress. In the $(\sigma_1, \sigma_2, \tau_{12})^i$ space, this inequality defines an ellipsoid or a sphere, depending on the Y_1 , Y_2 and Y_{12} values, whose surface is the border between elastic and plastic regimes.

Assuming that the different yield stresses can be computed from the elastic theory using an isotropic strain threshold, one obtains

$$\left(\frac{\sigma_1^i}{C_{11}}\right)^2 - \frac{\sigma_1^i \sigma_2^i}{C_{11}^2} + \left(\frac{\sigma_2^i}{C_{22}}\right)^2 + \left(\frac{\tau_{12}^i}{C_{12}}\right)^2 < \varepsilon_0^2, \tag{12}$$

where ε_0 is the strain threshold.

Computing the β_1^i , β_2^i and β_{12}^i values is then done as follows:

- Knowing the different stress components as well as the different parameters $(\varepsilon_0, Q_{11},...)$, one can check whether or not the previous inequality has been violated, i.e., whether the current location in the $(\sigma_1, \sigma_2, \tau_{12})^i$ space is outside the ellipsoid or not.
- If it is outside, the three yield parameters, β_1^i , β_2^i and β_{12}^i , can be found by taking the orthogonal projection of the current location in the $(\sigma_1, \sigma_2, \tau_{12})^i$ space onto the ellispoid, which is equivalent to finding the shortest distance between the point and the ellispoid. This procedure is described in the next section.
- Once the yield parameters are known, one can use them in Eq. (8) in order to compute the stress changes.

3.5 Projection onto an ellipsoid

Here we find the location of a point in the $(\sigma_1, \sigma_2, \tau_{12})^i$ space that would lie onto the ellispoid defined by Eq. (11), and minimize the distance with the known stress values. We first explain the solution for an "ideal" ellispoid (in which there is no $\sigma_1\sigma_2$ product), and then explain the change of variable that is necessary to use this solution in our case.

3.5.1 The case of an ideal ellispoid

Let $x = (x_1, x_2, x_3)$ being a point onto the surface of an ellispoid whose equation is

$$g(x) = \left(\frac{x_1}{a_1}\right)^2 + \left(\frac{x_2}{a_2}\right)^2 + \left(\frac{x_3}{a_3}\right)^2 - 1 = 0.$$
 (13)

Let $y = (y_1, y_2, y_3)$ being a point located outside the ellipsoid. The squared distance function between x and y is

$$d(x) = (x_1 - y_1)^2 + (x_2 - y_2)^2 + (x_3 - y_3)^2.$$
(14)

Our goal is to find x that minimizes d(x) while satisfying g(x) = 0.

At the minimal or maximal distance, the normal vectors to d(x) and q(x) are parallel, i.e.,

$$\nabla d(x) = \lambda \nabla g(x),\tag{15}$$

where λ is a Lagrange multiplier (a scalar). The previous equation can be expressed for all variables,

$$\frac{\mathrm{d}d(x)}{\mathrm{d}x_1} = \lambda \frac{\mathrm{d}g(x)}{\mathrm{d}x_1},$$

$$\frac{\mathrm{d}d(x)}{\mathrm{d}x_2} = \lambda \frac{\mathrm{d}g(x)}{\mathrm{d}x_2},$$

$$\frac{\mathrm{d}d(x)}{\mathrm{d}x_3} = \lambda \frac{\mathrm{d}g(x)}{\mathrm{d}x_3},$$
(16)

which gives the expression of x as a function of y, λ and the ellipsoid parameters,

$$x_{1} = \frac{y_{1}}{1 - \lambda/a_{1}^{2}},$$

$$x_{2} = \frac{y_{2}}{1 - \lambda/a_{2}^{2}},$$

$$x_{3} = \frac{y_{3}}{1 - \lambda/a_{3}^{2}}.$$
(17)

These solutions can then be plugged into the constraint g(x) = 0, which gives

$$g(\lambda) = \frac{y_1^2}{a_1^2 (1 - \lambda/a_1^2)^2} + \frac{y_2^2}{a_2^2 (1 - \lambda/a_2^2)^2} + \frac{y_3^2}{a_3^2 (1 - \lambda/a_3^2)^2} - 1 = 0,$$
(18)

where all parameters but λ are known. The value for λ can then be found using a numerical optimization procedure such as, e.g., a Newton algorithm where λ is obtained through an iterative approach,

$$\lambda_{k+1} = \lambda_k - \frac{g(\lambda_k)}{g'(\lambda_k)},\tag{19}$$

where k is the Newton iteration subscript, and

$$g'(\lambda_k) = \frac{\mathrm{d}g(\lambda_k)}{\mathrm{d}\lambda} = \frac{2y_1^2}{a_1^4(1 - \lambda_k/a_1^2)^3} + \frac{2y_2^2}{a_2^4(1 - \lambda_k/a_2^2)^3} + \frac{2y_3^2}{a_3^4(1 - \lambda_k/a_3^2)^3}.$$
 (20)

One can start this procedure with $\lambda_0 = 0$ as an intial guess, which should avoid the search of a maxima instead of a minima. Once λ is known, we use its value to compute x_1 , x_2 and x_3 . Please keep in mind that this procedure has to be repated for every wall layer.

3.5.2 Adaptation to our case

In our case, and for a given wall layer, the ellipsoid equation at the minimal distance with the point $(\sigma_1, \sigma_2, \tau_{12})$ is

$$\left(\frac{\beta_1}{Y_1}\right)^2 - \frac{\beta_1 \beta_2}{Y_1^2} + \left(\frac{\beta_2}{Y_2}\right)^2 + \left(\frac{\beta_{12}}{Y_{12}}\right)^2 - 1 = 0, \tag{21}$$

which can be compared to Eq. (13). We must use a change of variable to be able to use the previous solution to our case. Recognizing that our ellipsoid underwent a rotation of angle α around the τ_{12} axis (writing the ellipsoid equation in a matrix form, $\sigma^T \overline{\overline{A}} \sigma = 1$, one can see that in our case the matrix is not diagonal and that the τ_{12} axis is an eigenvector), we can write

$$\beta_1 = \cos(\alpha)x_1 + \sin(\alpha)x_2,$$

$$\beta_2 = -\sin(\alpha)x_1 + \cos(\alpha)x_2,$$

$$\beta_{12} = x_3,$$
(22)

and

$$y_1 = \cos(\alpha)\sigma_1 - \sin(\alpha)\sigma_2,$$

$$y_2 = \sin(\alpha)\sigma_1 + \cos(\alpha)\sigma_2,$$

$$y_3 = \sigma_{12}.$$
(23)

Injecting these expressions into our ellispoid equation and identifying the different terms with respect to Eq. (13), one obtains

$$\frac{1}{a_1^2} = \frac{\cos^2(\alpha)}{Y_1^2} + \frac{\sin^2(\alpha)}{Y_2^2} + \frac{\sin(\alpha)\cos(\alpha)}{Y_1^2},$$

$$\frac{1}{a_2^2} = \frac{\sin^2(\alpha)}{Y_1^2} + \frac{\cos^2(\alpha)}{Y_2^2} - \frac{\cos(\alpha)\sin(\alpha)}{Y_1^2}$$

$$\frac{1}{a_3^2} = \frac{1}{Y_{12}^2},$$
(24)

and

$$\tan(2\alpha) = \frac{Y_2^2}{Y_2^2 - Y_1^2}. (25)$$

Thus the full solution can be computed.

Note that the different procedures described in this document have to be computed for all layers, as the MF angle is different from one layer to another and these layers have not been deposited at the same time. Only the mechanical properties in the MF frame (the coefficient of the rigidity matrix $\overline{\overline{Q}}$) as well as the ellipsoid parameters are the same for all layers.

The procedure to compute the yield threshold in the MF frame will be encapsulated in a function that outputs the yield threshold for all wall layers and that takes as inputs the elastic properties and all wall layer stresses.

References

- S. Abrate. The mechanics of short fiber-reinforced composites: a review. Rubber chemistry and technology, 59(3):384–404, 1986.
- C. Fuchs, D. Bhattacharyya, and S. Fakirov. Microfibril reinforced polymer–polymer composites: Application of Tsai-Hill equation to pp/pet composites. *Composites science and technology*, 66(16):3161–3171, 2006.
- J. Halpin and J. Kardos. The Halpin-Tsai equations: a review. Polymer Engineering & Science, 16(5):344–352, 1976.
- S. Mortazavian and A. Fatemi. Effects of fiber orientation and anisotropy on tensile strength and elastic modulus of short fiber reinforced polymer composites. *Composites part B: engineering*, 72:116–129, 2015.
- G. Song, C. Lancelon-Pin, P. Chen, J. Yu, J. Zhang, L. Su, M. Wada, T. Kimura, and Y. Nishiyama. Time-dependent elastic tensor of cellulose nanocrystal probed by hydrostatic pressure and uniaxial stretching. The Journal of Physical Chemistry Letters, 12(15):3779–3785, 2021.
- S. W. Tsai. Strength characteristics of composite materials. NASA Washington, DC, 1965.