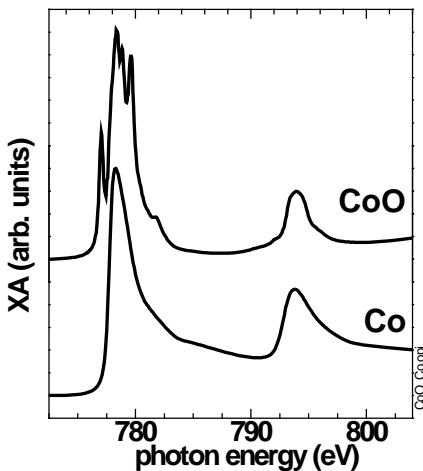
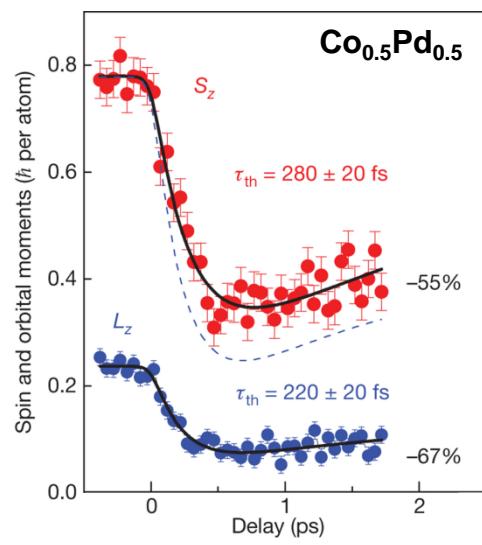


# MAGNETIC SPECTROSCOPY

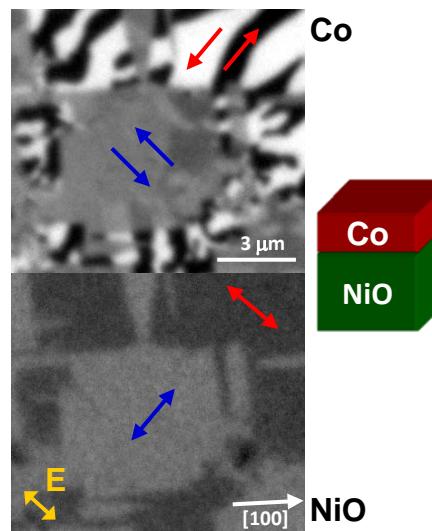
Elke Arenholz, Advanced Light Source



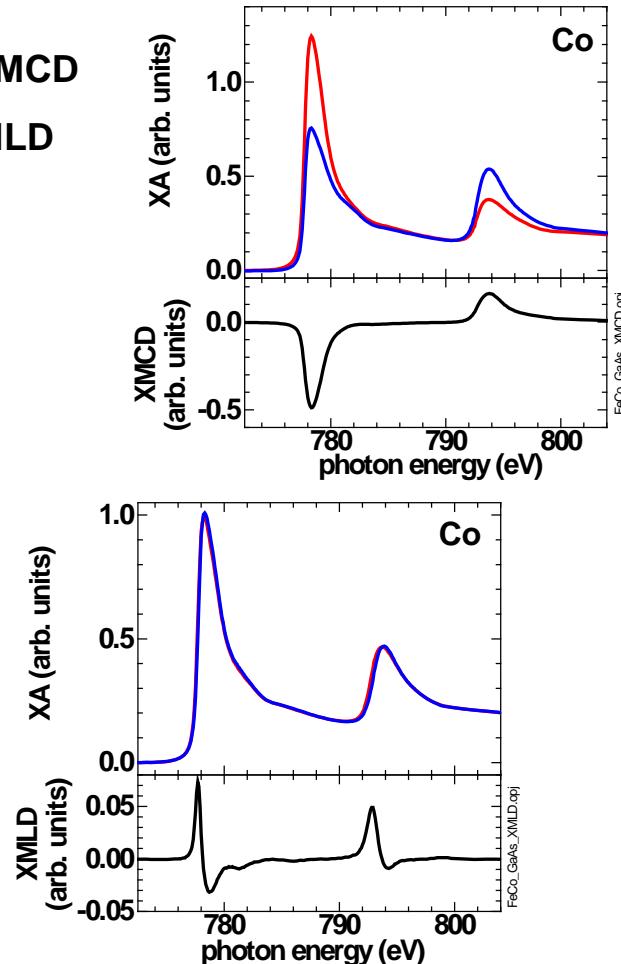
- + X ray absorption, XA
- + X ray magnetic circular dichroism, XMCD
- + X ray magnetic linear dichroism, XMLD
- + X ray magnetic microscopy
- + Magnetization Dynamics



C. Boeglin et al., Nature 465, 458 (2011)

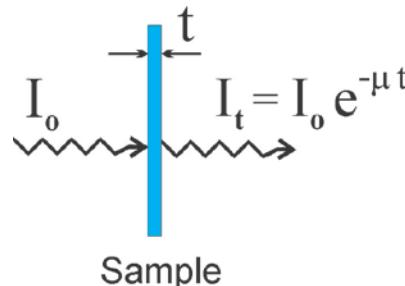


E. Arenholz et al.,  
Appl. Phys. Lett. 93, 162506 (2008)



U.S. DEPARTMENT OF  
**ENERGY**  
Office of Science

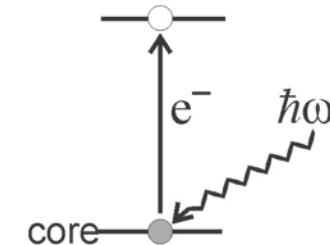




**Experiment/Measurement:**  
Reduction in x ray flux transmitted through a sample.

### X-ray absorption:

- + Electrons excited from core shells to unoccupied valence states through the absorption of a photon determined by energy and angular momentum conservation

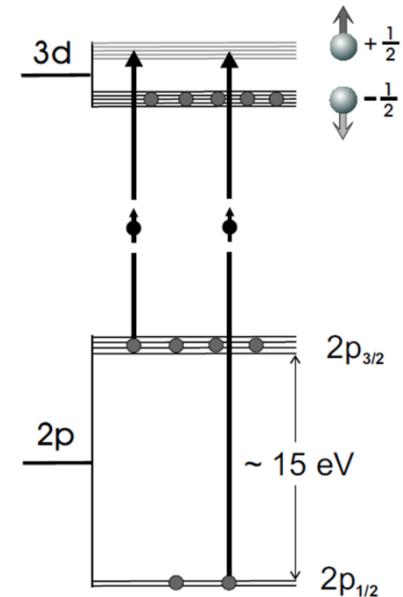


### Simplest model: One electron picture

- + Photon transfers its energy and momentum to core electron
- + Core electron excited into unoccupied electronic state.
- + However: Not directly excited electrons also influenced by electron excitation, i.e. hole in core shell

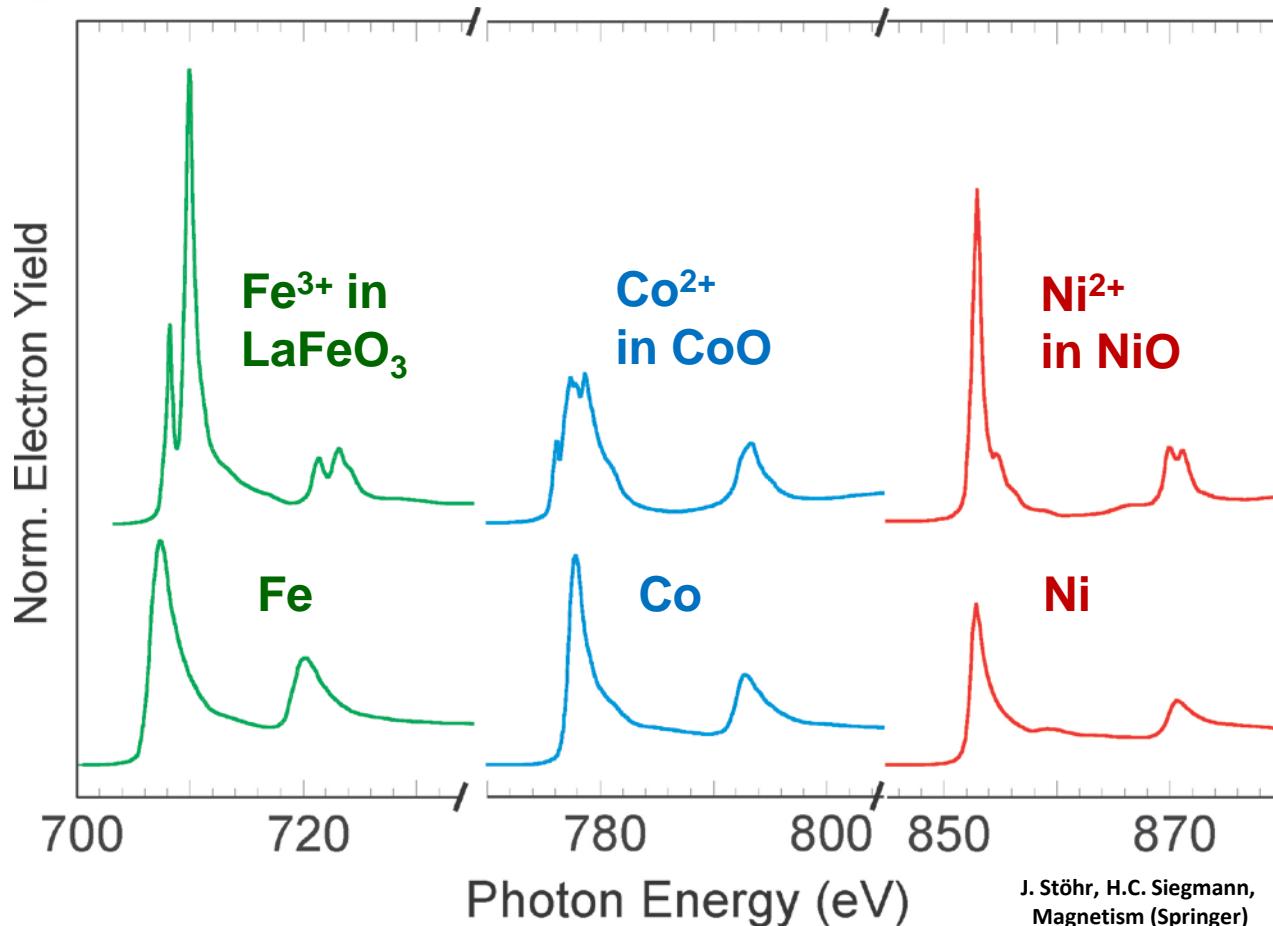
Configuration model, e.g. *L* edge absorption :

- + Atom is excited from ground/initial state configuration,  $2p^63d^n$  to excited/final state configuration,  $2p^53d^{n+1}$
- + Omission of all full subshells (spherical symmetric)
- + Takes into account correlation effects
  - in the ground state as well as
  - in the excited state
- + Leads to multiplet effects/structure



J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

# X-RAY ABSORPTION

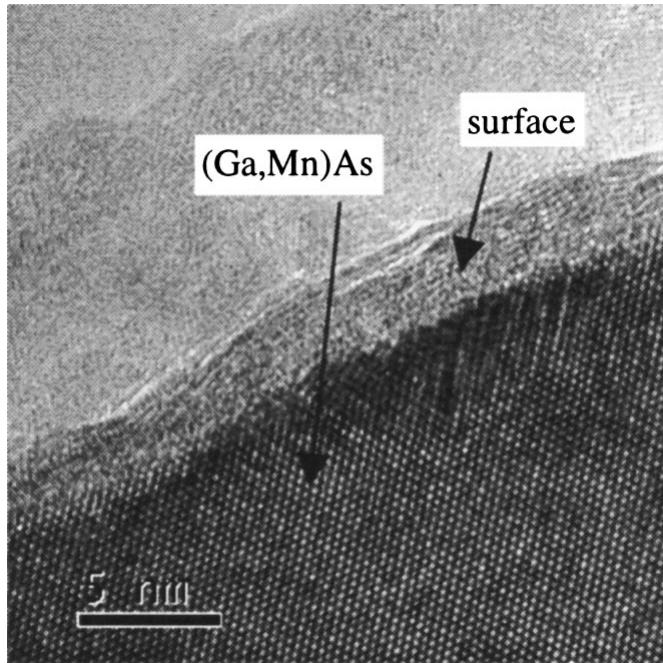


XA provides

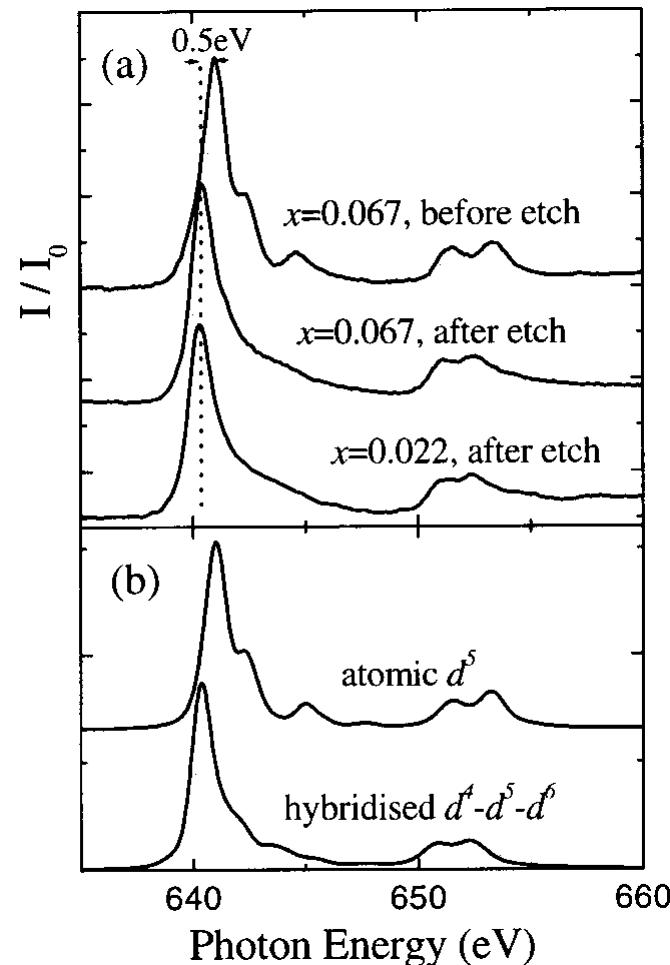
- + elemental specificity
- + sensitivity to valence shell properties,  
i.e. valence state and lattice site symmetry

J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

# SURFACE EFFECTS IN (Ga,Mn)As

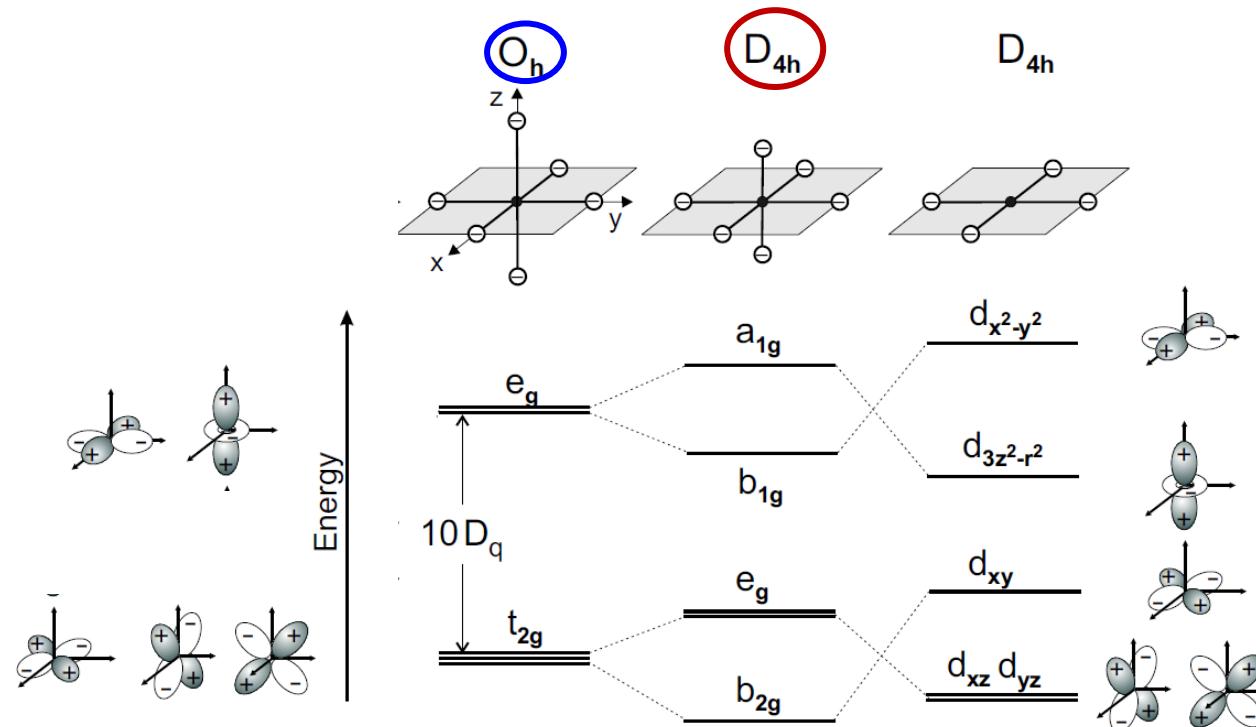


- + As grown/before etch:
  - Multiplet structure characteristic of MnO
- + After removal of the surface layer:
  - Multiplet structure is less pronounced
  - Spectrum shifted to 0.5 eV lower photon energy.
- + Comparison with calculated spectra:
  - localized Mn ground state for the untreated sample
  - hybridized ground state after etching.



K. Edmonds *et al.*,  
Appl. Phys. Lett. **84**, 4065(2004)

# SENSITIVITY TO SITE SYMMETRY: $Ti^{4+} L_{3,2}$



+ Electric dipole transitions:  $d^0 \rightarrow 2p^5 3d^1$

+ Crystal field splitting  $10Dq$  acting on  $3d$  orbitals:

J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

### Octahedral symmetry:

$e$  orbitals towards ligands  $\rightarrow$  higher energy

$t_2$  orbitals between ligands  $\rightarrow$  lower energy

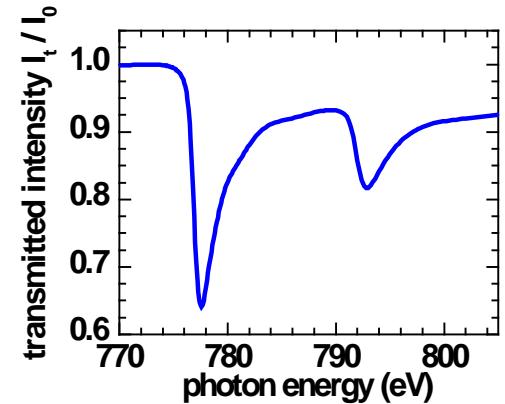
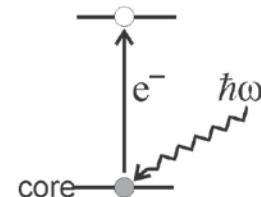
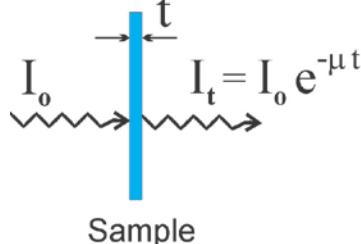
### Tetragonal symmetry:

$e$  orbitals  $\rightarrow b_2 = d_{xy}$ ,  $e = d_{yz}$ ,  $d_{yz}$

$t_2$  orbitals  $\rightarrow b_1 = d_{x^2-y^2}$ ,  $a_1 = d_{3z^2-r^2}$

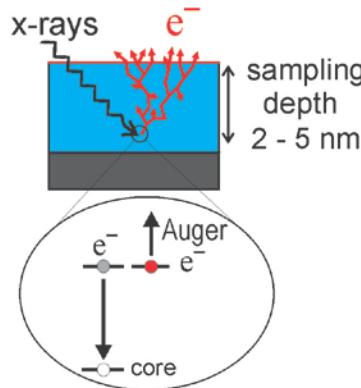
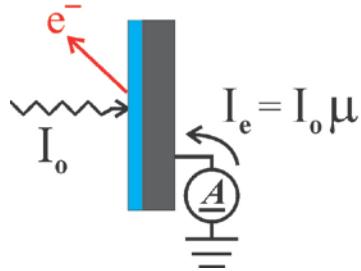
# X-RAY ABSORPTION – MEASUREMENTS

## Transmission

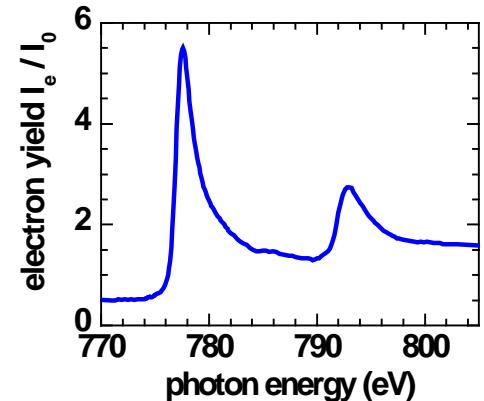


photons absorbed

## Electron Yield



J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

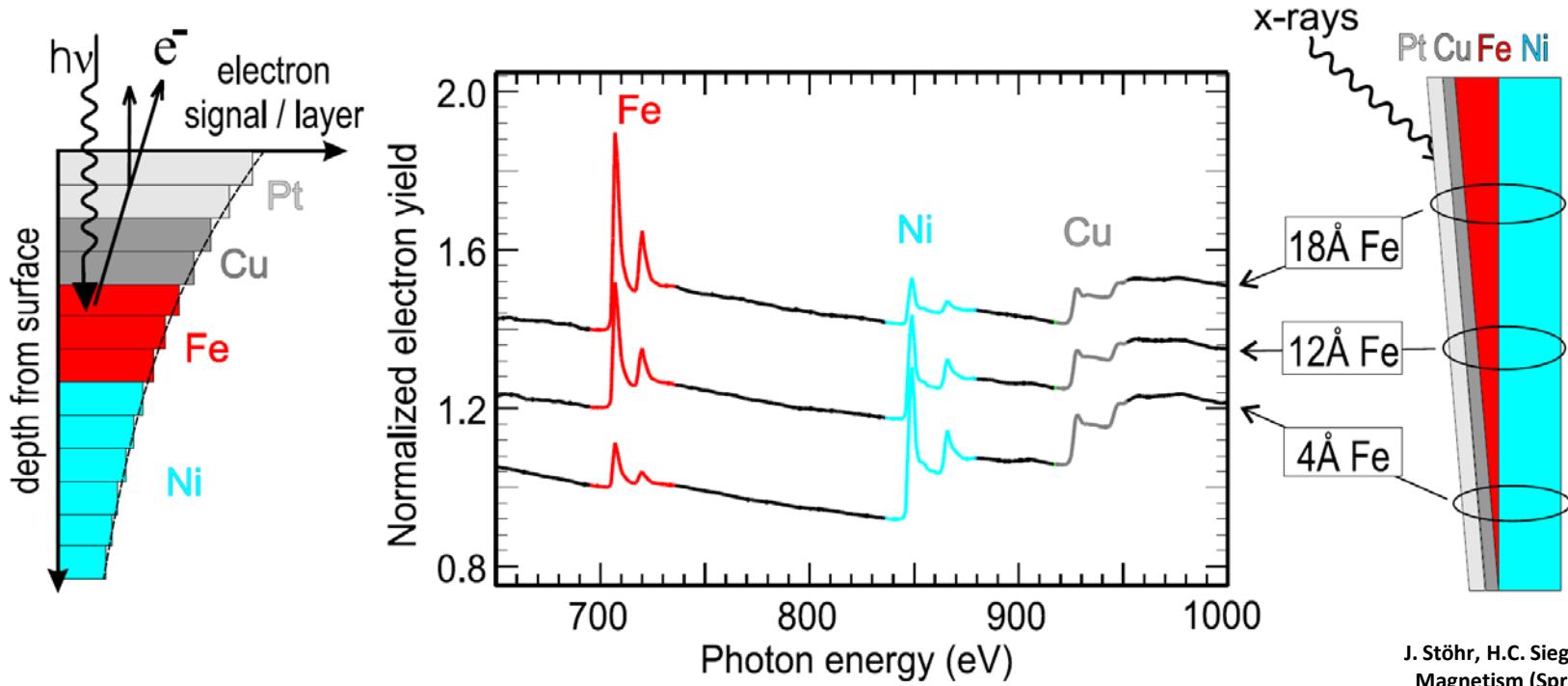


electrons generated

## Electron yield:

- + Absorbed photons create core holes that are filled predominantly by Auger electron emission
- + Auger electrons create low-energy secondary electron cascade through inelastic scattering
- + Emitted electrons  $\propto$  probability of Auger electron creating  $\propto$  absorption probability

# SAMPLING DEPTH OF ELECTRON YIELD



J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

- + Electron sample depth: 2nm in Fe, Co, Ni  
 ⇒ 60% of the electron yield originates from the topmost 2nm
- + X ray absorption length: 500nm before the absorption edge  
 20nm at the  $L_3$  edge

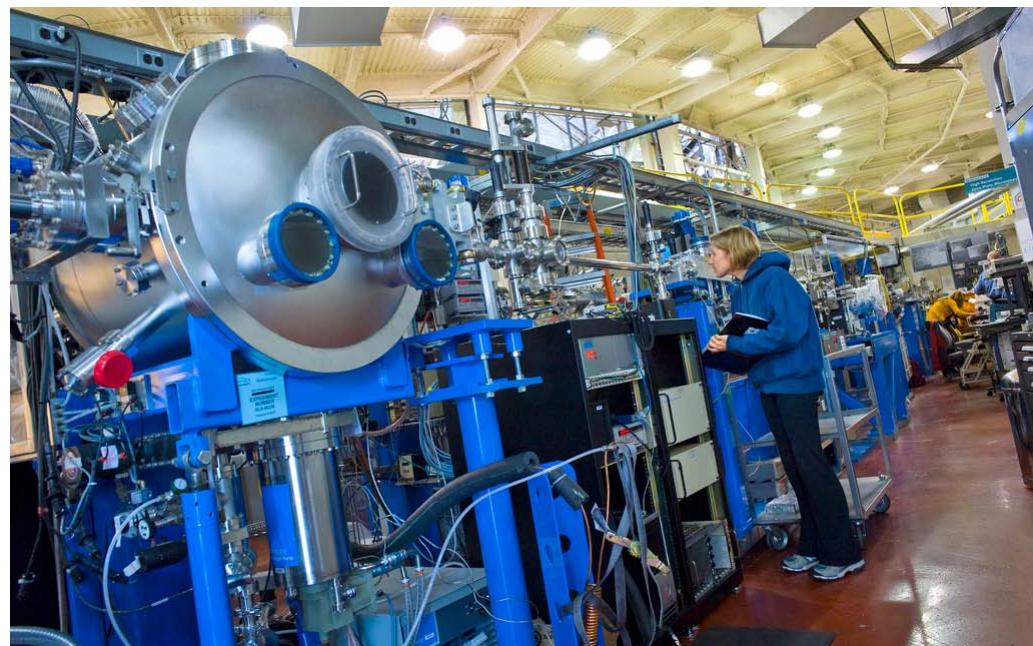
# PHOTON SOURCES AND MONOCHROMATORS

## Advanced Light Source

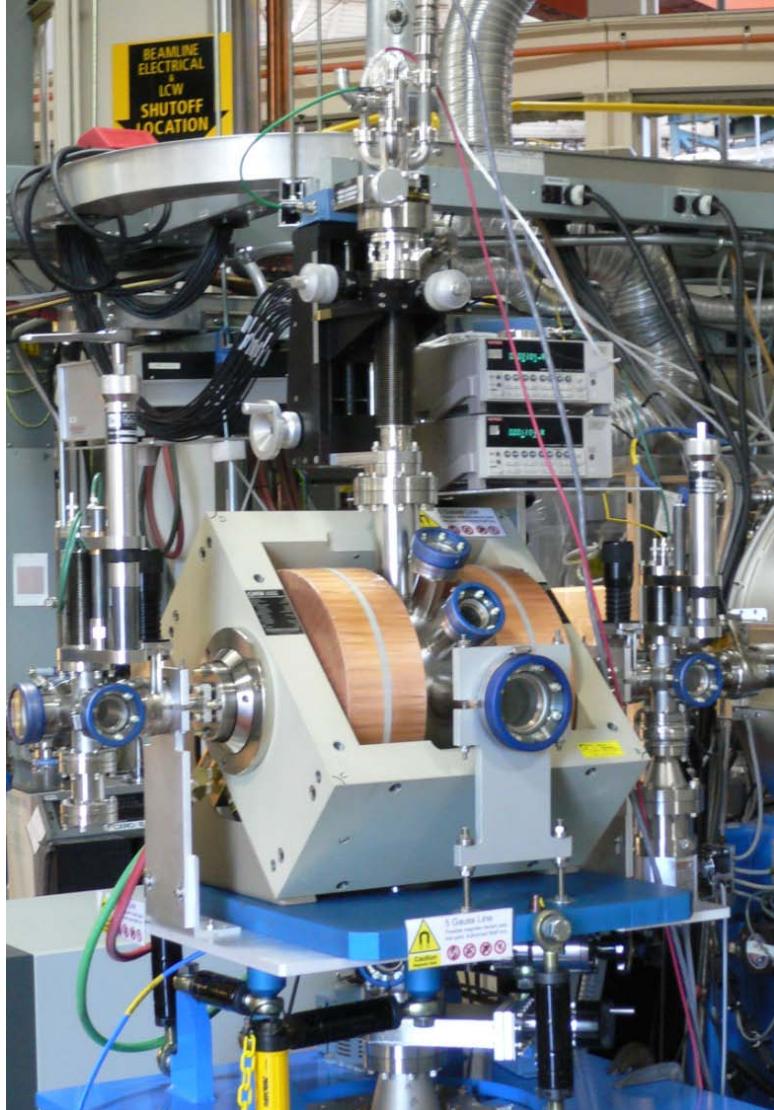


- + Tunable photon source in the soft x ray range, 200-2000eV, i.e. undulator or bend magnet, at synchrotron.
- + Beamlines/Monochromators provide photons with well defined characteristics:
  - tunable energy/wavelength
  - fixed polarization: (variable) linear, circular, elliptical, ...

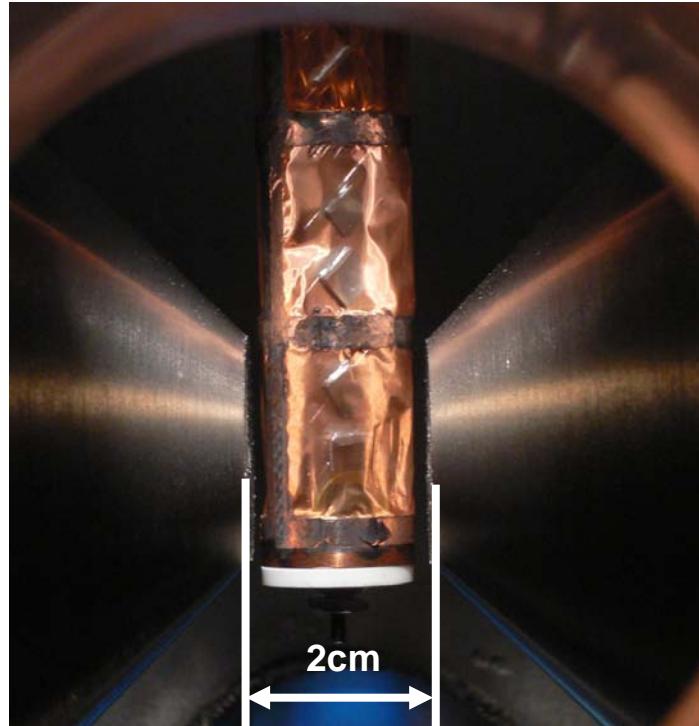
BL6.3.1



# ENDSTATIONS

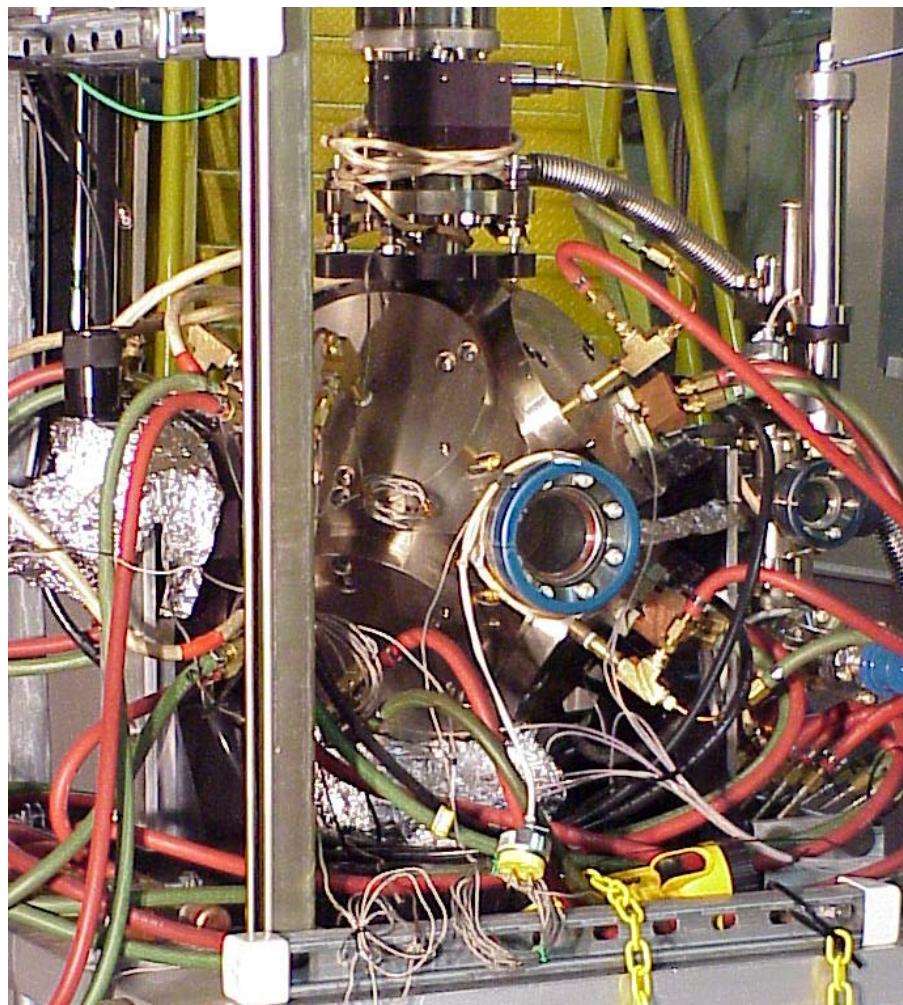


- + Endstations provide well defined sample environments for the interaction with photons:
  - precisely defined experimental geometries
  - sample temperature
  - external magnetic and electric fields
  - ...



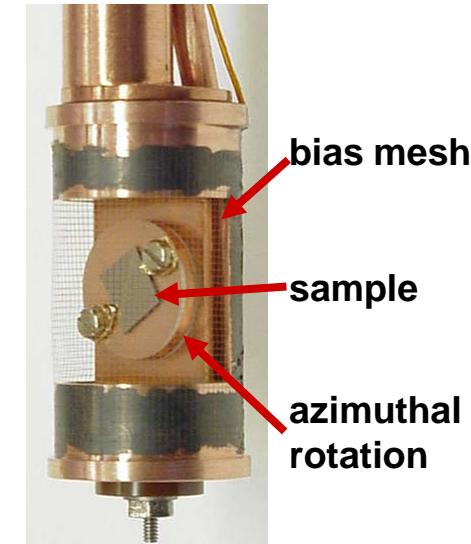
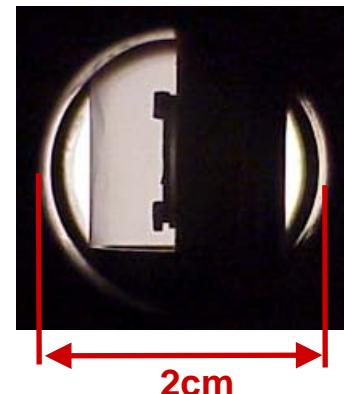
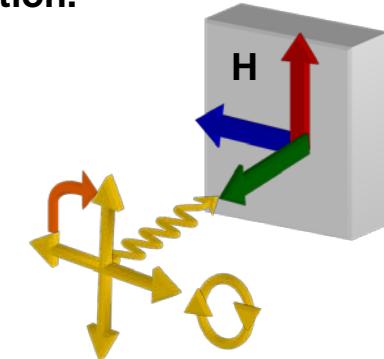
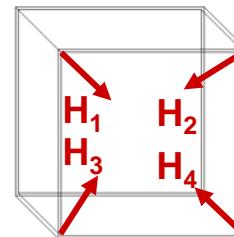
2T magnet at  
ALS BL6.3.1

# ENDSTATIONS

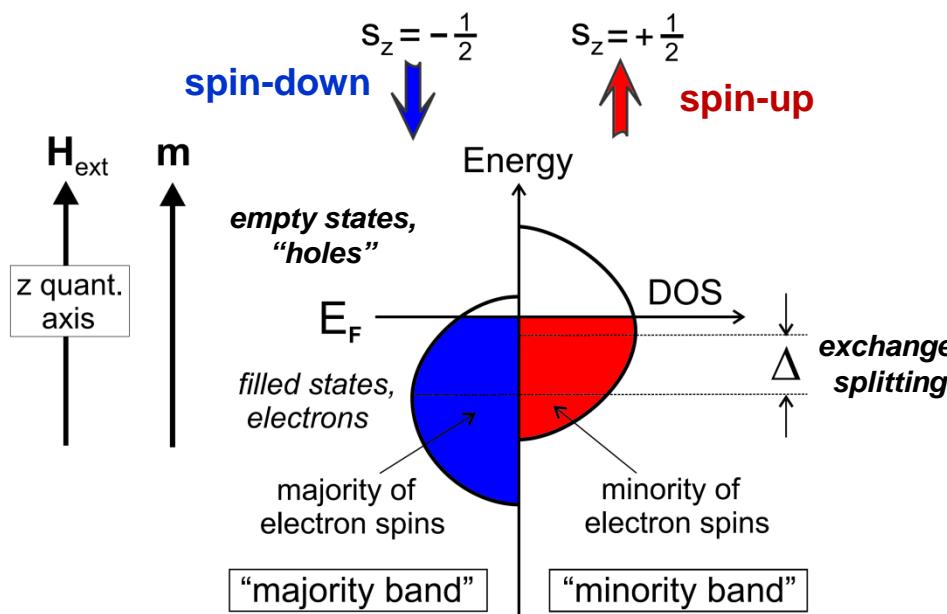


Vector magnet at ALS BL4.0.2

+ Magnetic fields in arbitrary directions obtained through superposition of fields generated by 4 dipole pairs in octahedral configuration.



# STONER MODEL FOR FERROMAGNETIC TRANSITION METALS



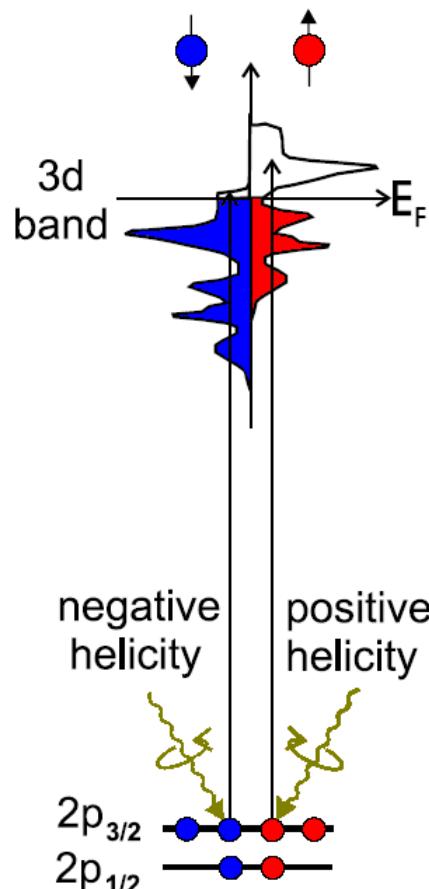
J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

**3d shell**

- + Magnetic moments in Fe, Co, Ni are well described by the Stoner model: d-bands containing up and down spins shifted relative to each other by “exchange splitting”
- + Spin- up and spin-down bands filled according to Fermi statistics.
- + Magnetic moment  $|m|$  determined the difference in number of electrons in majority and minority bands

$$|m| = \mu_B (n_e^{\text{maj}} - n_e^{\text{min}})$$

## TWO-STEP MODEL OF XMCD



Photoelectrons excited from  $2p_{3/2}$ ,  $2p_{1/2}$  to 3d states

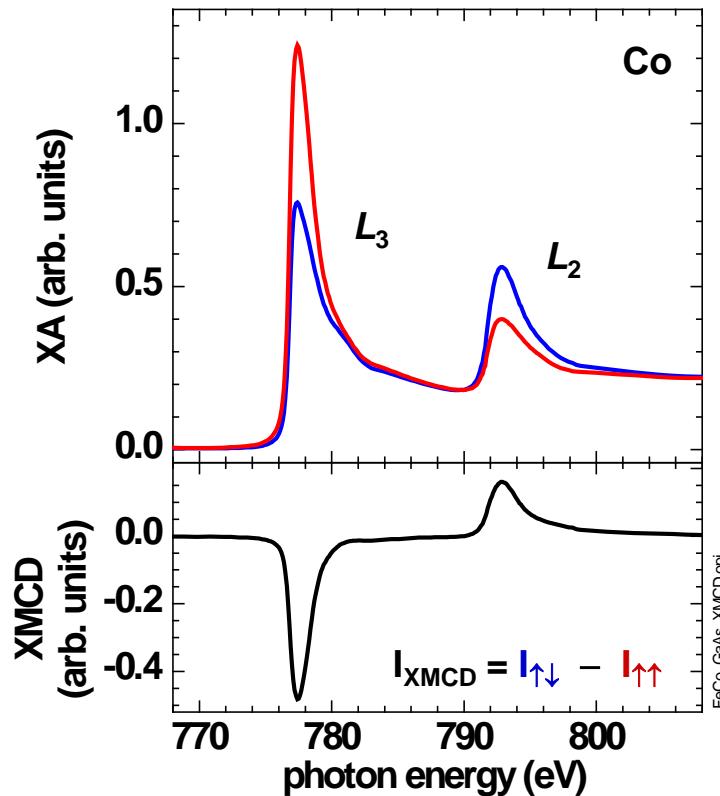
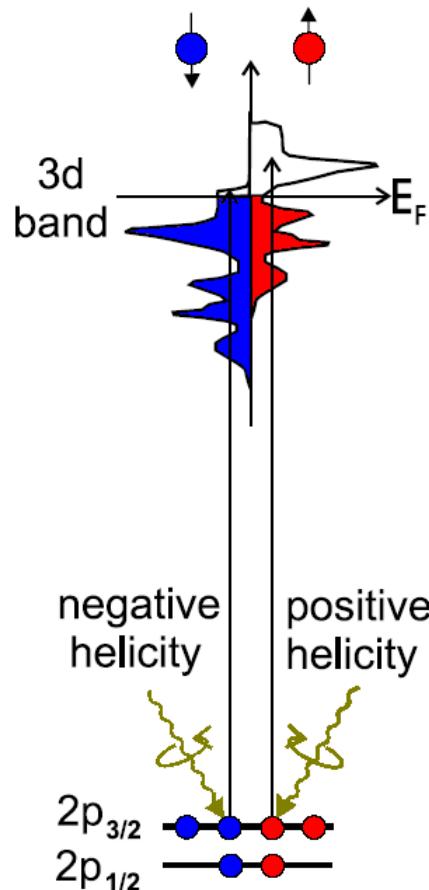
### First step:

- + Excitation of electron from  $2p$  states by absorption of circularly polarized x rays.
- + Note: Dipole operator does not act on the spin and  
⇒ No spin flips during excitation.
- + Conservation of angular momentum  
⇒ transfer of angular momentum ( $\pm\hbar$ ) from photon to electron
- + Spin-orbit coupling: Angular momentum of photon transferred  
in part to electron spin  
⇒ Excited photoelectrons are spin polarized

### Second step:

- + Unequal spin-up and spin-down populations  
determines spin or orbital momentum of possible excitations

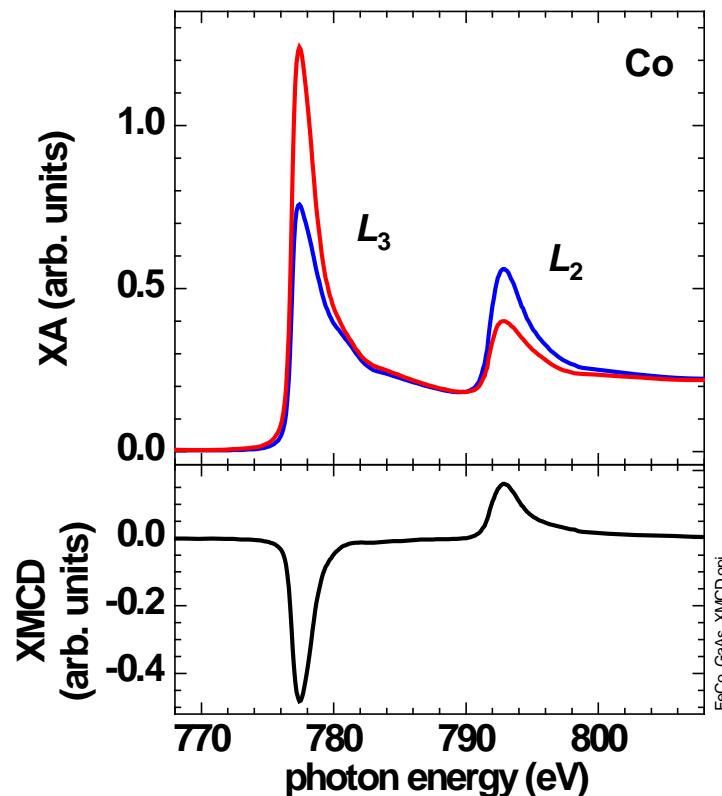
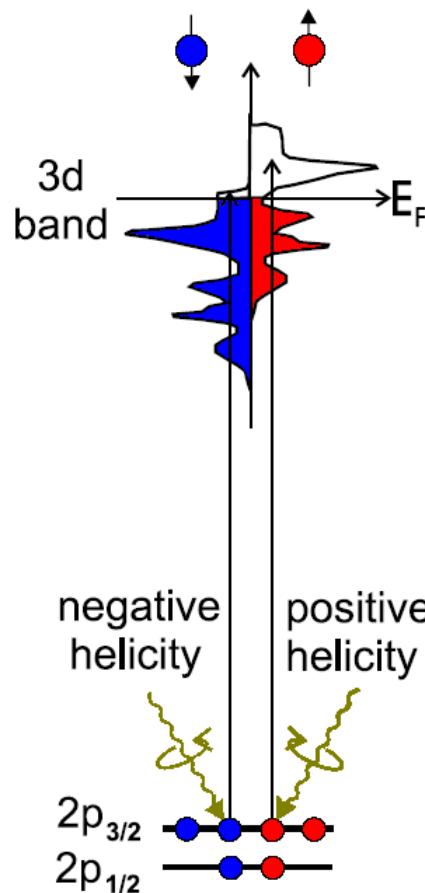
J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

TWO-STEP MODEL OF XMCD

Magnitude of the dichroism effect depends on

- + degree of circular photon polarization,  $P_{\text{circ}}$
  - + angle  $\theta$  between photon angular momentum,  $L_{\text{ph}}$  and magnetic moment,  $m$
  - + expectation value of 3d magnetic moment
- $$I_{\text{XMCD}} \propto P_{\text{circ}} \langle m \rangle \cos \theta$$
- + XMCD allows studying ferri- and ferromagnets.

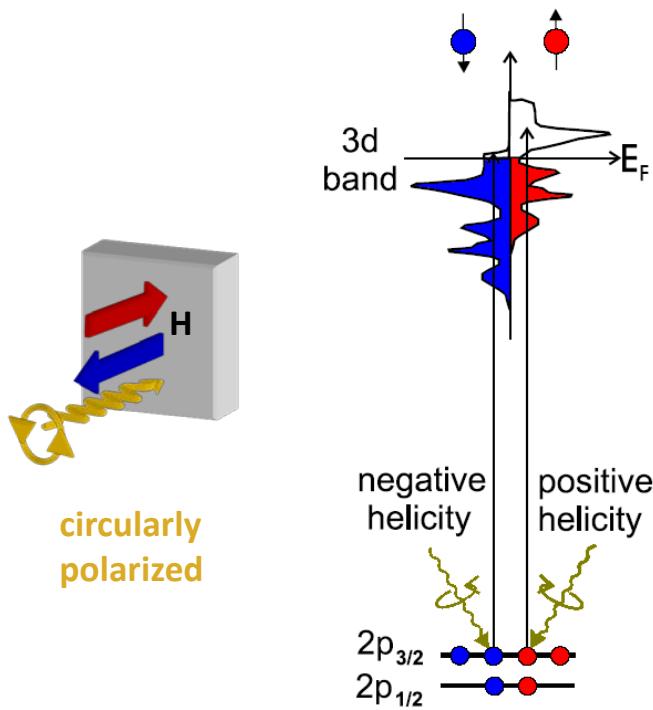
J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

TWO-STEP MODEL OF XMCD

- +  $2p_{3/2}$  and  $2p_{1/2}$  have opposite spin orbit coupling ( $\mathbf{l}+\mathbf{s}$ ,  $\mathbf{l}-\mathbf{s}$ )  
⇒ Spin polarization and XMCD have opposite sign at two edges
- + Spin polarization opposite for x rays with opposite helicity, i.e. photon spin,  $\pm\hbar$   
⇒ XMCD reverses sign with polarization
- + Reversing the x ray polarization is equivalent to reversing magnetization/ spin direction

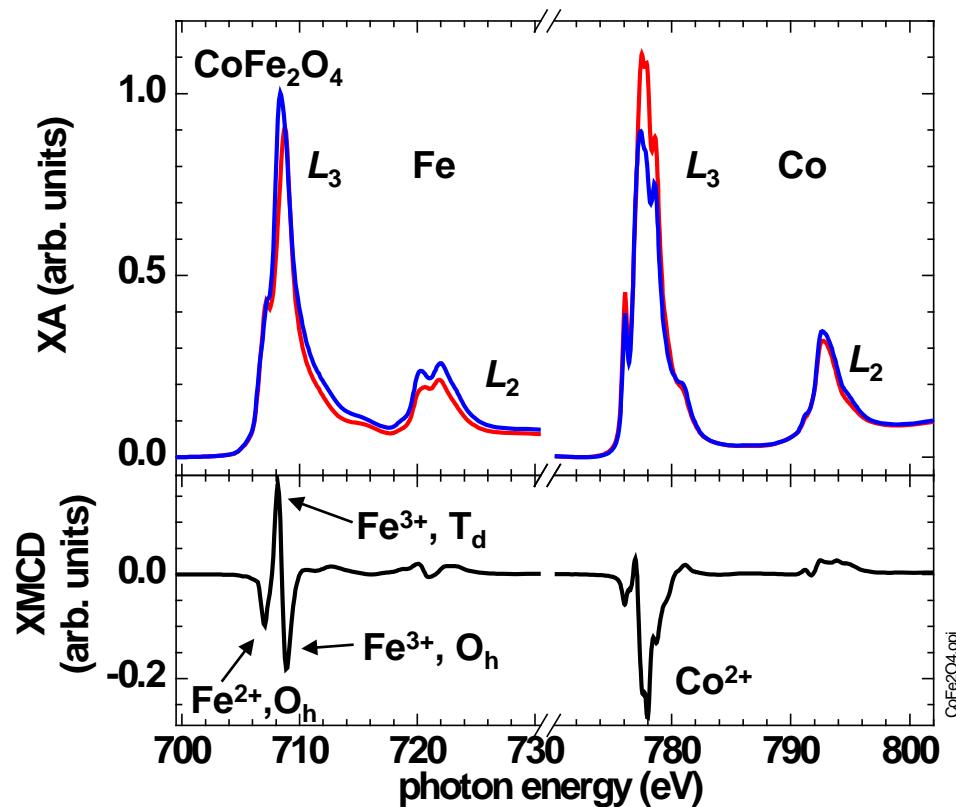
J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

# X RAY MAGNETIC CIRCULAR DICHROISM (XMCD)



circularly  
polarized

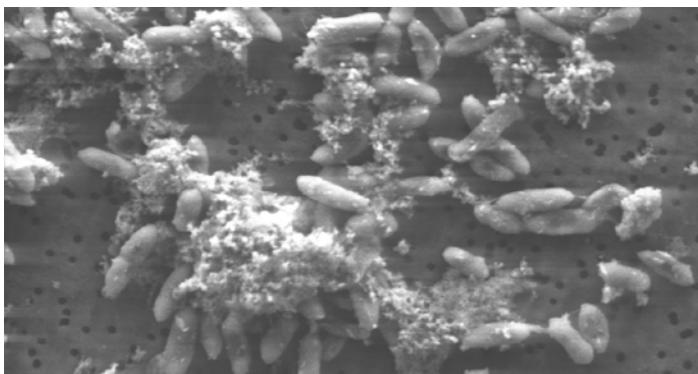
J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)



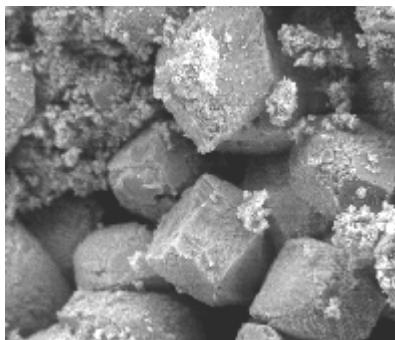
- + XMCD provides magnetic information resolving elements Fe, Co, ...
- valence states:  $\text{Fe}^{2+}$ ,  $\text{Fe}^{3+}$ , ...
- lattice sites: octahedral,  $\text{O}_h$ , tetrahedral,  $\text{T}_d$ , ...

# CHARACTERISTICS OF MAGNETIC BIONANOSPINELS

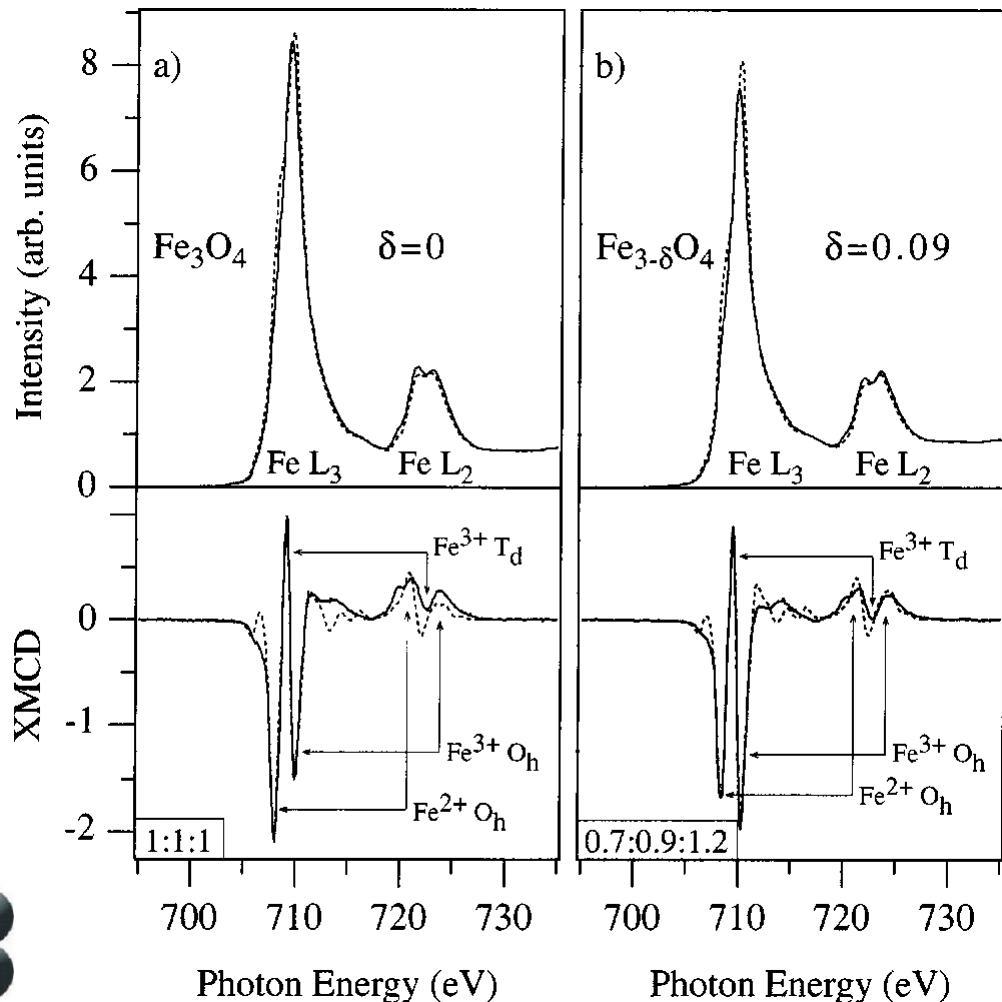
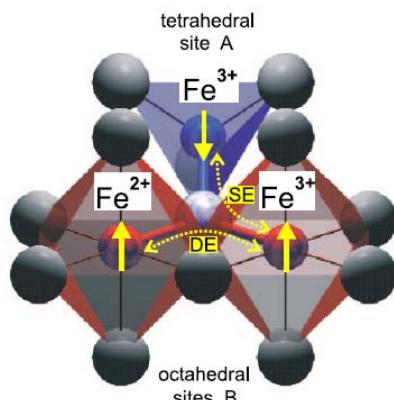
+ **Geobacter sulfurreducens bacteria form magnetite via extracellular reduction of amorphous Fe(III)-bearing minerals**



V. Cocker et al.,  
Eur. J. Mineral. 19, 707–716 (2007)



The Department of Geology and Geophysics,  
University of Wisconsin-Madison

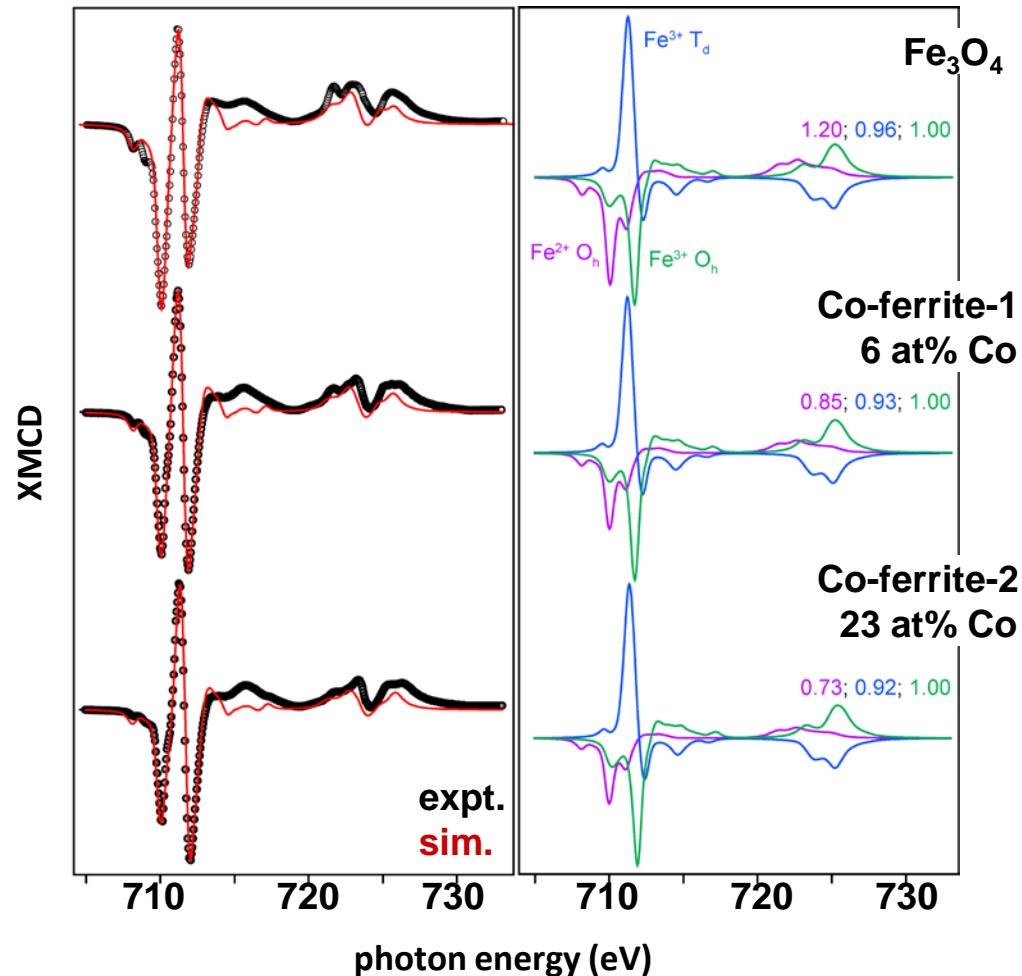
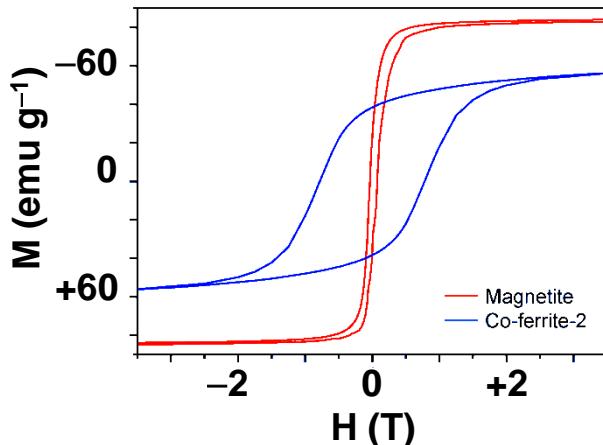


P. Morrall et al.,  
Phys. Rev. B 67, 214408 (2003)

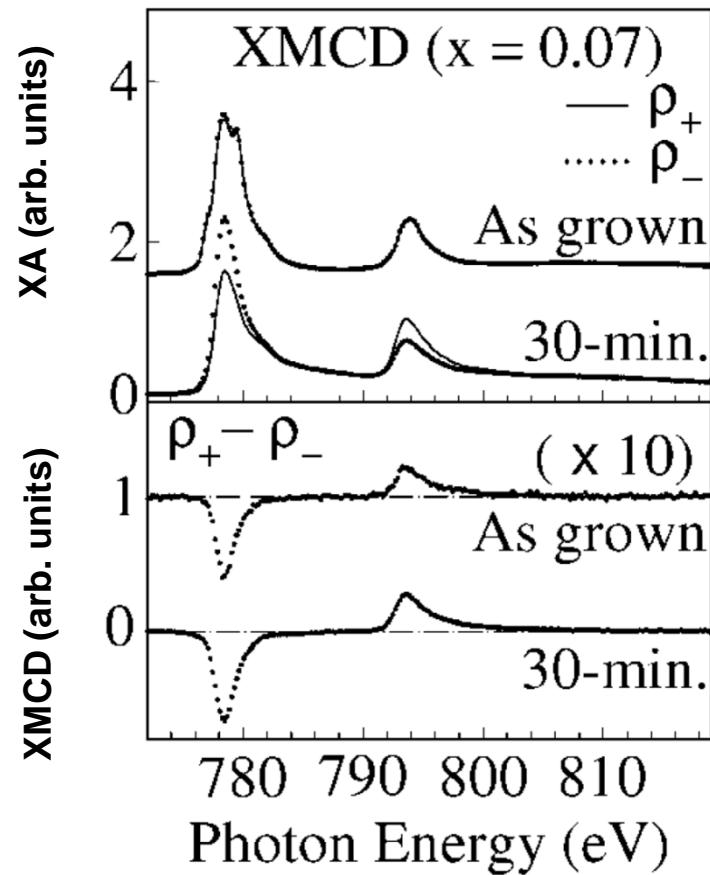
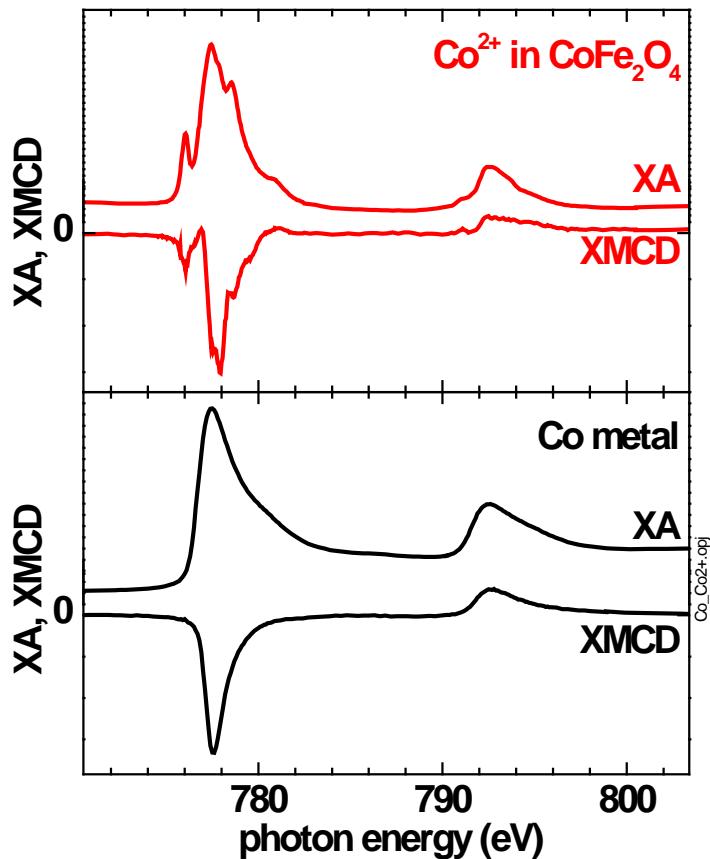
# CHARACTERISTICS OF MAGNETIC BIONANOSPINELS

Magnetite and Co ferrites produced from Co(II) containing Fe(III)-oxyhydroxides using metal-reducing bacterium (*Geobacter sulfurreducens*)

- + Up to 23 at% Co<sup>2+</sup> incorporated (compared to 1 at% using magnetotactic bacteria)
- + Co<sup>2+</sup> in Fe<sup>2+</sup> O<sub>h</sub> sites
- + 10fold increase in magnetic anisotropy



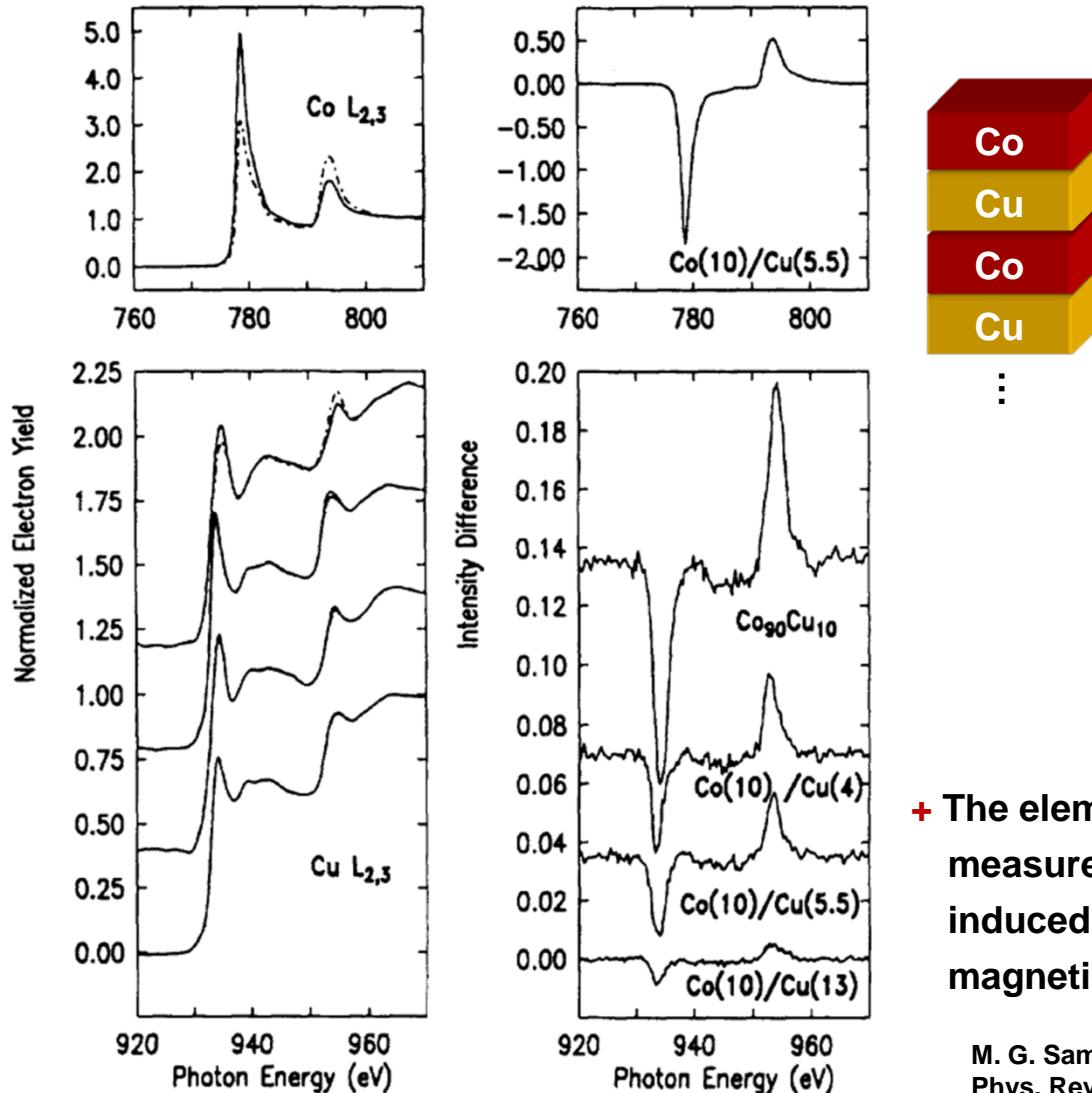
V. Cocker *et al.*,  
Eur. J. Mineral. **19**, 707–716 (2007)

Co-DOPED TiO<sub>2</sub>

- + Comparing XMCD spectra with model compounds and/or theoretical multiplet calculations allows
- ⇒ Identifying the contributions to the magnetic phase of a system.

J.-Y. Kim et al.,  
Phys. Rev. Lett. **90**, 017401 (2003)

# INDUCED MOMENTS AT Co/Cu INTERFACES



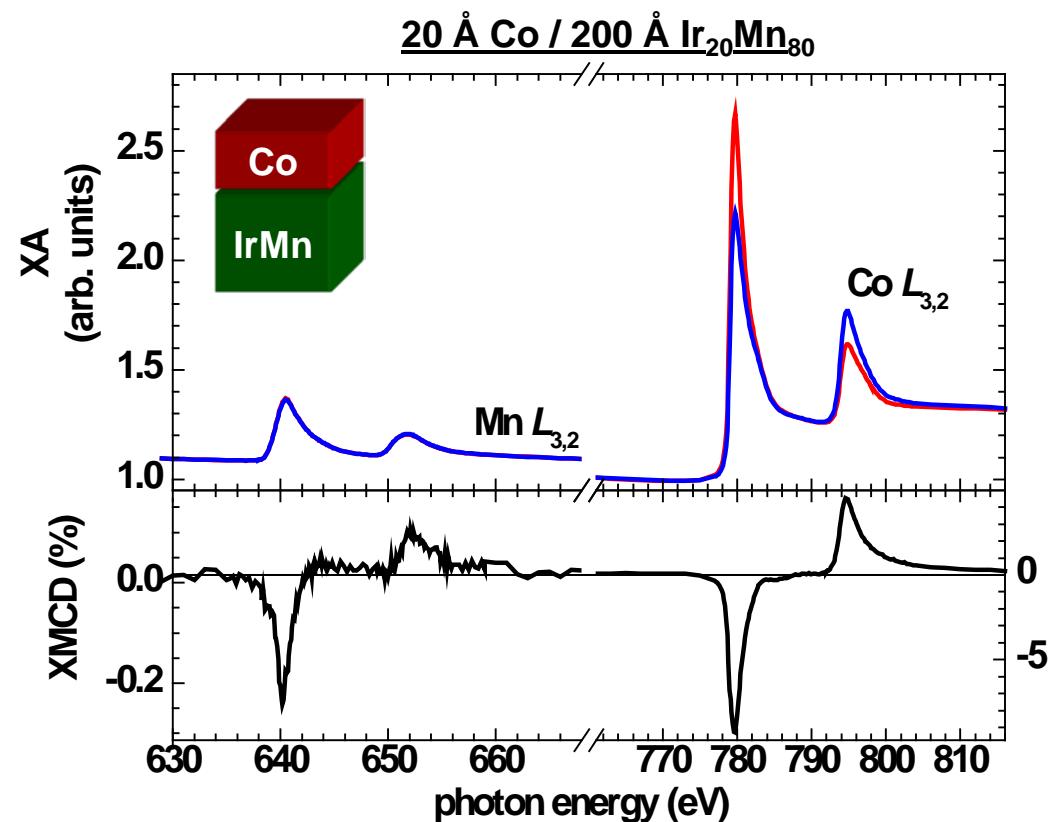
+ The element-specificity makes XMCD measurements an ideal tool to determine induced moments at interfaces between magnetic and non-magnetic elements.

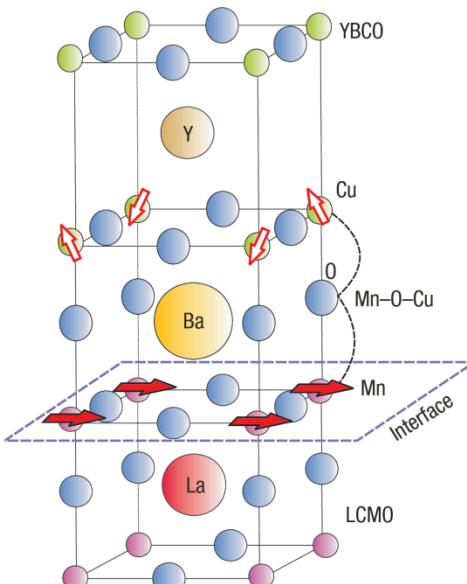
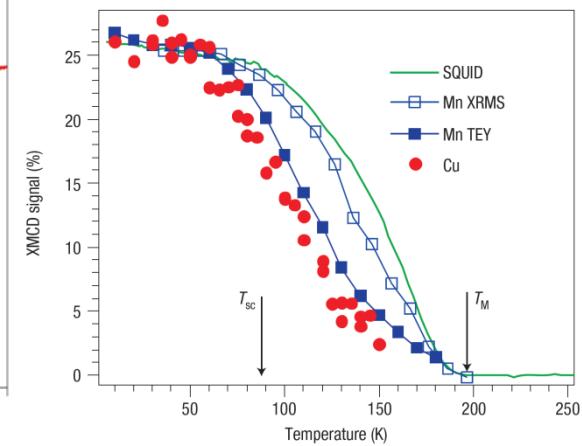
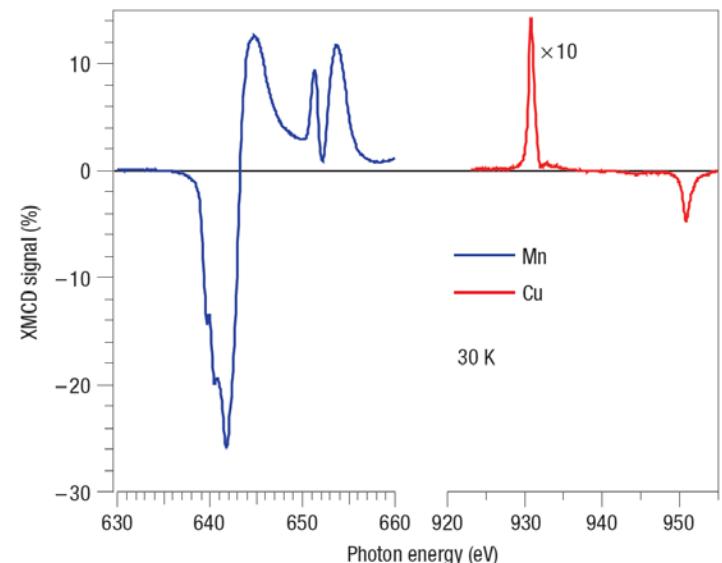
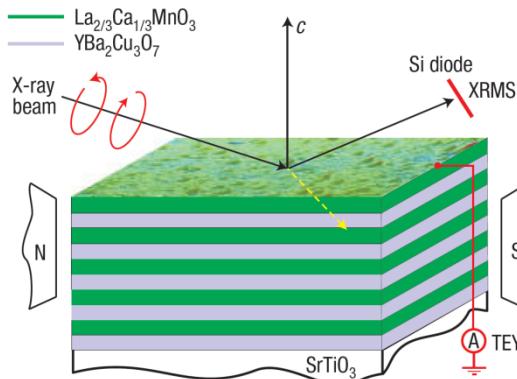
M. G. Samant *et al.*,  
Phys. Rev. Lett. 72, 1112 (1994)

# ANTIFERROMAGNET/FERROMAGNETIC INTERFACES

ALS

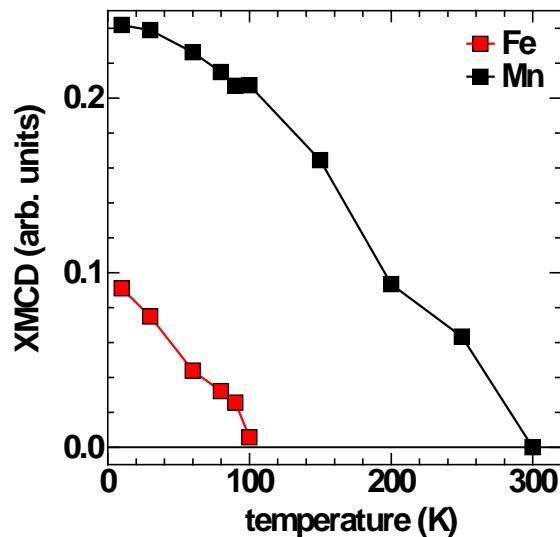
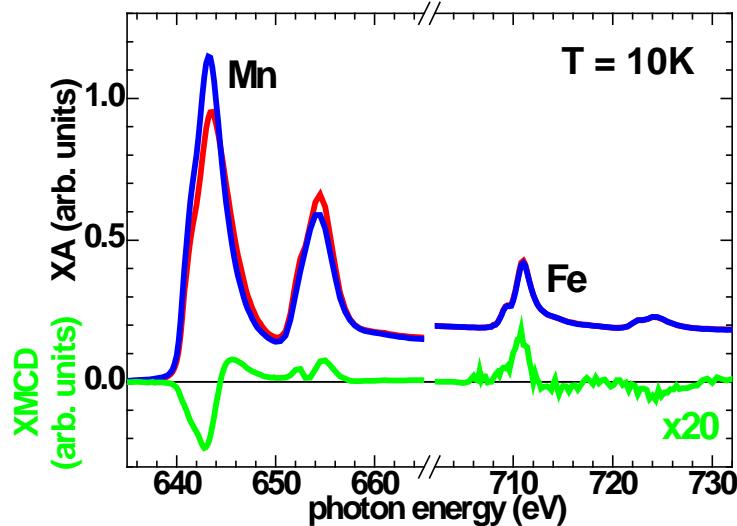
- + The weak Mn XMCD signal indicates uncompensated Mn at the Co/IrMn interface.
- + The same sign of XMCD signal for Co and Mn and indicates parallel coupling.
- + The nominal thickness of uncompensated interface moments is  $(0.5 \pm 0.1)$ ML for Co/Ir<sub>20</sub>Mn<sub>80</sub>.





J. Chakalian *et al.*,  
Nature Phys. **2**, 244 (2006)

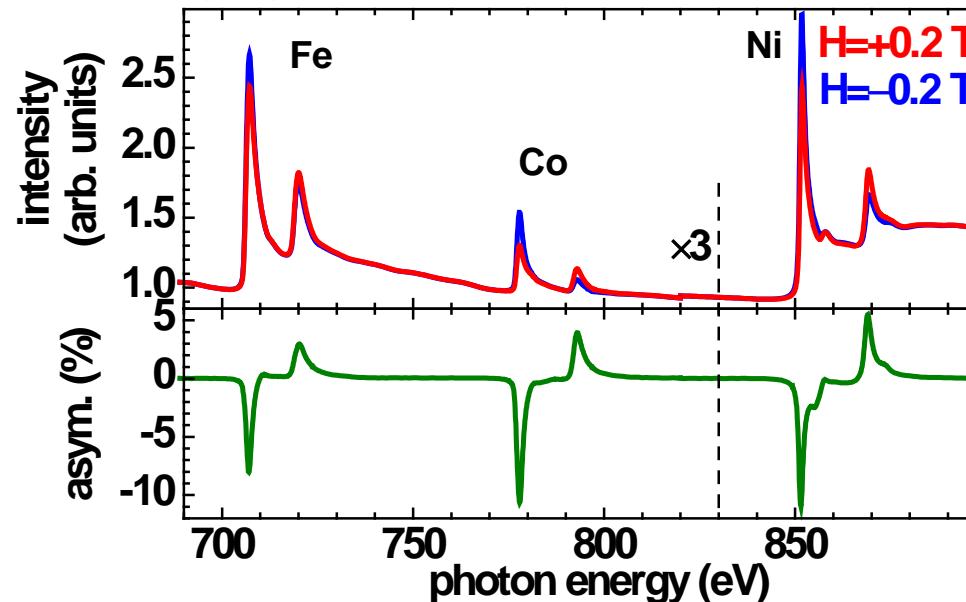
- + **LCMO : significant Mn  $L_{3,2}$  XMCD at  $T = 30\text{K}$**   
**ferromagnetic transition  $\sim 180\text{ K}$**
- + **YBCO: Weak Cu  $L_{3,2}$  XMCD**  
 $\Leftrightarrow$  **net ferromagnetic polarization on Cu**  
**i.e. presence of uncompensated induced magnetic moment in the YBCO layer close to LCMO interface.**
- + **opposite sign of Cu and Mn XMCD**  
 $\Leftrightarrow$  **antiparallel orientation of Cu and Mn moments**

$\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3 / \text{BiFeO}_3$  INTERFACE

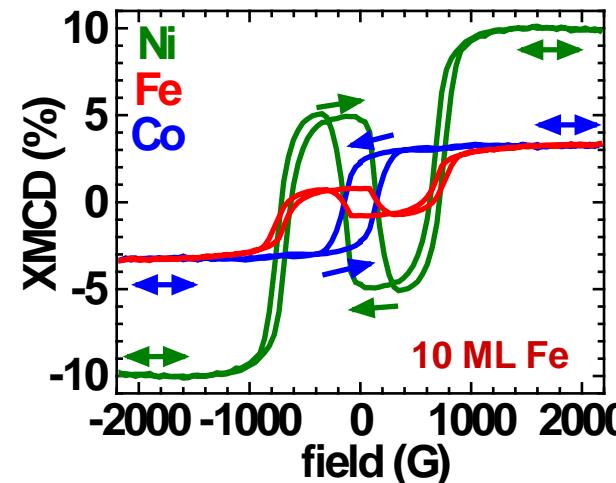
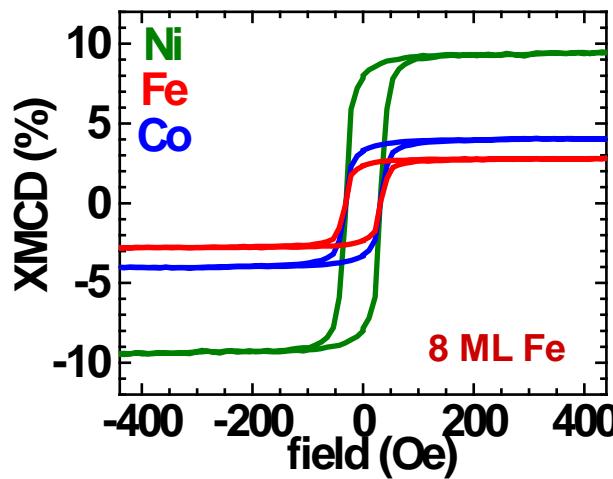
- +  $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$  : significant Mn  $L_{3,2}$  XMCD at T = 10K  
ferromagnetic transition ~300 K
- +  $\text{BiFeO}_3$ : Weak Fe  $L_{3,2}$  XMCD  
 ⇔ net ferromagnetic polarization on Fe  
 i.e. presence of uncompensated induced magnetic moment in the  $\text{BiFeO}_3$  layer close to  $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$  interface.
- + opposite sign of Fe and Mn XMCD  
 ⇔ antiparallel orientation of Fe and Mn moments
- + Transition temperature of the magnetic phase in  $\text{BiFeO}_3$  significantly lower than  $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$

P. Yu *et al.*,  
*Phys. Rev. Lett.* **105**, 027201 (2010)

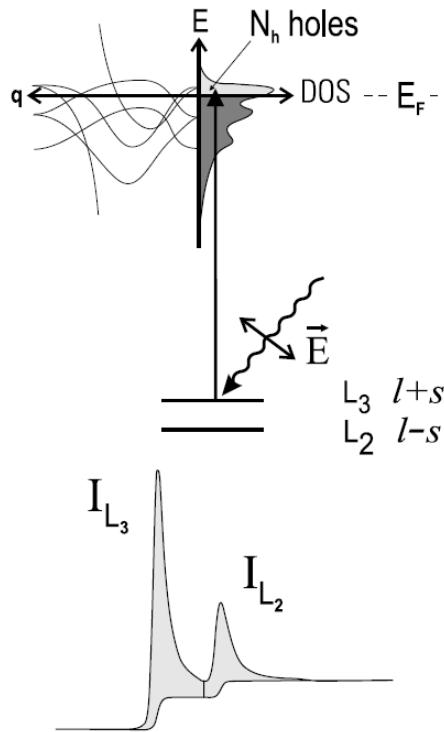
# ELEMENT-SPECIFIC MAGNETIZATION REVERSAL



- + Monitoring the field dependence of the XMCD signal
- Detailed information on magnetization reversal in complex magnetic hetero-structures



(a) d-Orbital Occupation



(b) Spin Moment

$$-\frac{1}{2} \downarrow \quad \uparrow +\frac{1}{2}$$

$$N^{\downarrow} \quad N^{\uparrow}$$

$$B > 0$$

$$A < 0$$

(c) Orbital Moment

$$-m_l \quad +m_l$$

$$N^{-m} \quad N^{+m}$$

$$B$$

$$A$$

**+ Theoretically derived sum rules correlate the XMCD spectra with the spin and orbital moment providing a unique tool for studying magnetic materials.**

$$N_h = \langle I_{L_3} + I_{L_2} \rangle / C \quad m_s = \mu_B \langle -A + 2B \rangle / C \quad m_L = -2\mu_B \langle A + B \rangle / 3C$$

J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

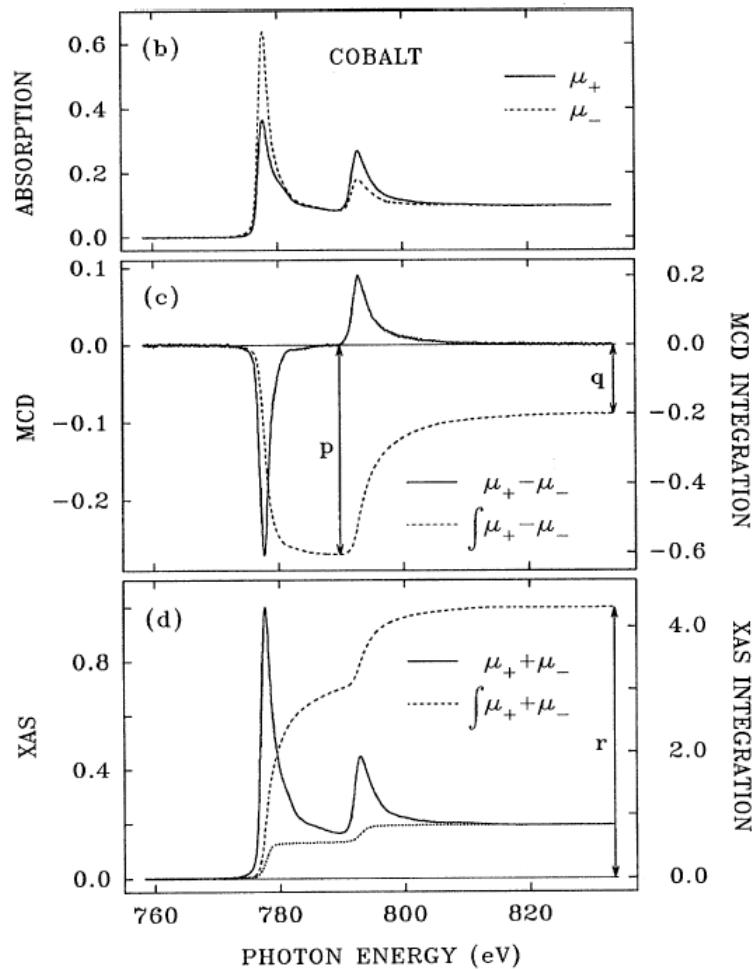
SUM RULES

+ Separation of spin and orbital moments requires very high quality data.

$$\frac{m_{\text{orb}}}{m_{\text{spin}}} = \frac{2q}{9p - 6q}$$

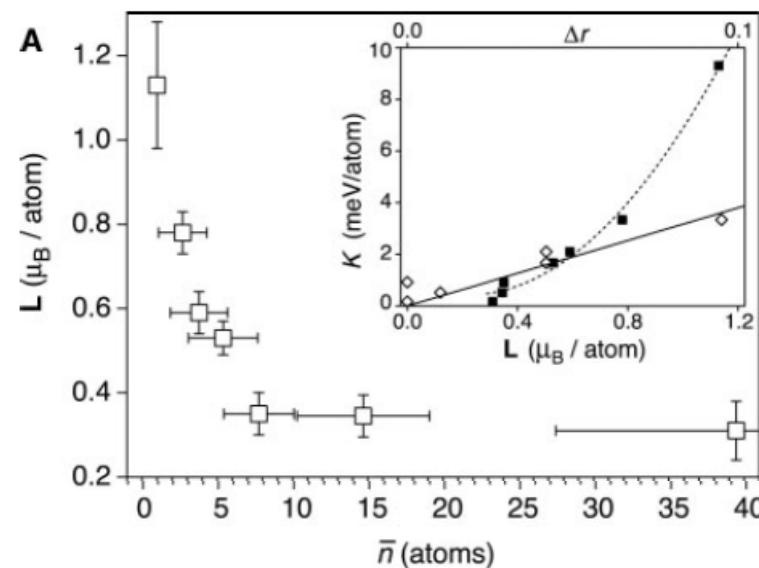
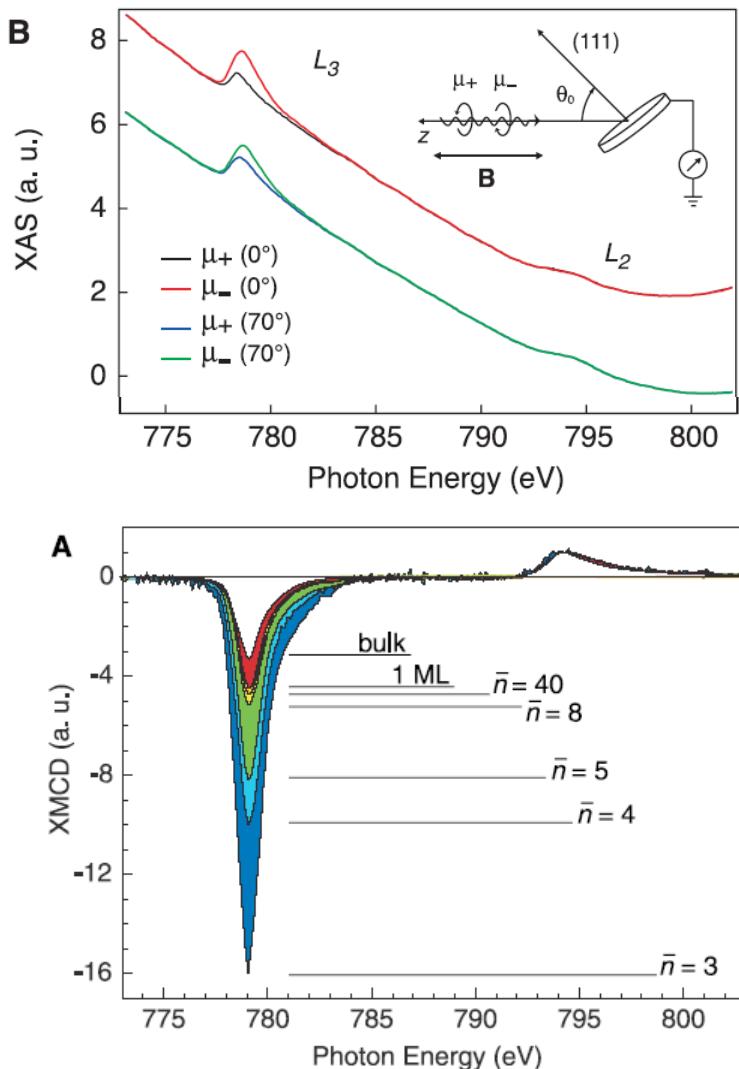
$$m_{\text{orb}} = -\frac{4q(10 - n_{3d})}{3r}$$

$$m_{\text{spin}} = -\frac{(6p - 4q)(10 - n_{3d})}{r}$$



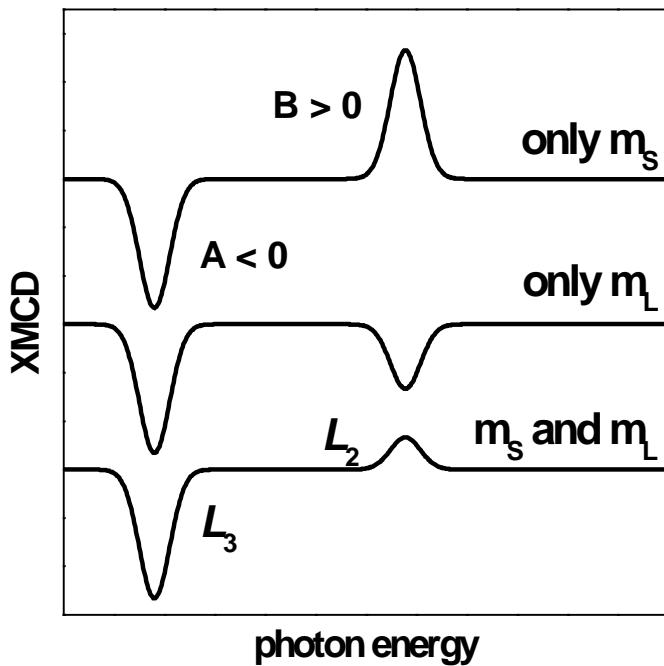
C.T. Chen *et al.*,  
Phys. Rev. Lett. **75**, 152(1995)

# ORBITAL MOMENT OF Co NANOPARTICLES



- + Strong variation of orbital and spin magnetic moment observable as change in  $L_3$  and  $L_2$  in the XMCD spectrum.
- + Co atoms and nanoparticles on Pt have enhanced orbital moments up to  $1.1 \mu_B$

P. Gambardella *et al.*,  
Science **300**, 1130 (2003)



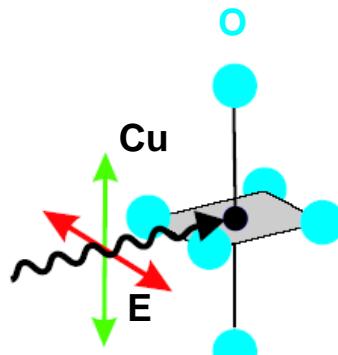
+ Spin and orbital moment only systems  
have distinct XMCD spectra:

$$m_L = -2\mu_B \langle A + B \rangle / 3C = 0 \quad \text{for } A = -B$$

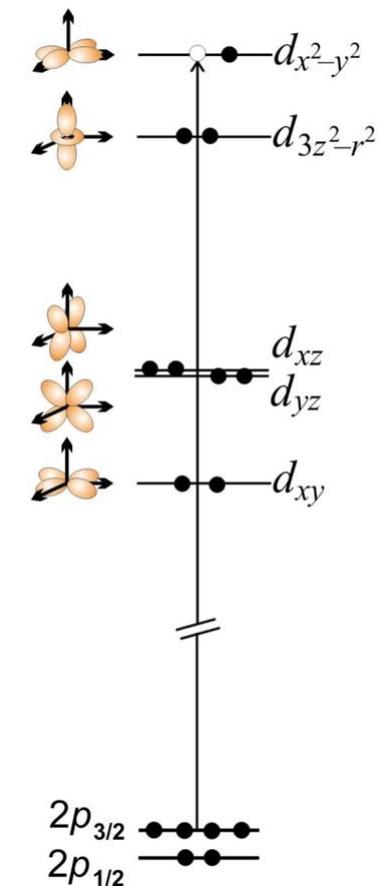
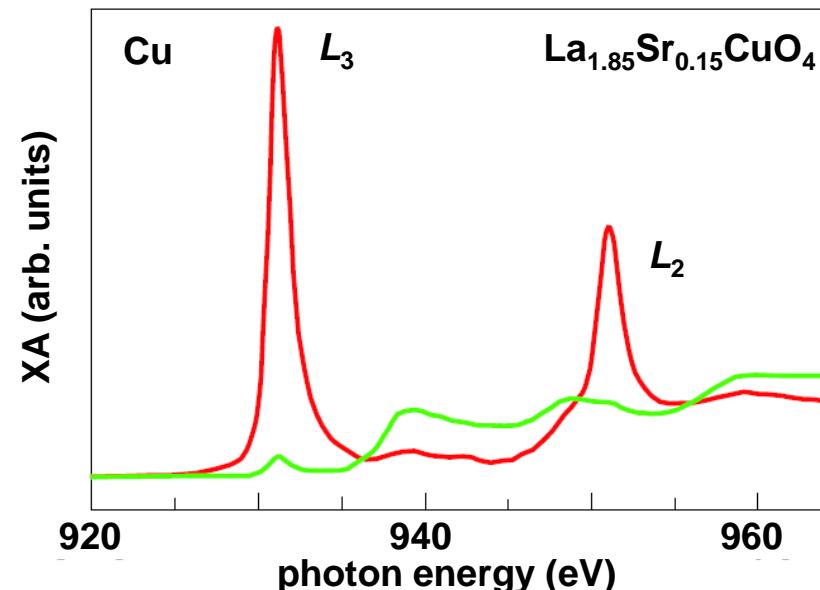
$$m_s = \mu_B \langle -A + 2B \rangle / C = 0 \quad \text{for } A = 2B$$

# X-RAY LINEAR DICHROISM

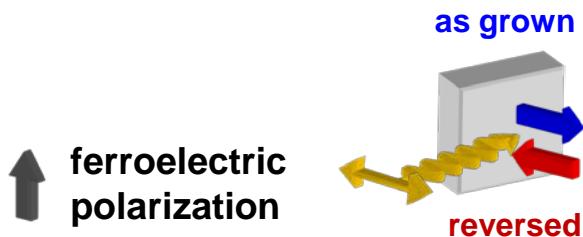
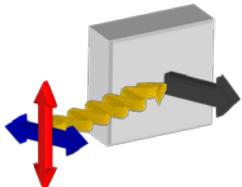
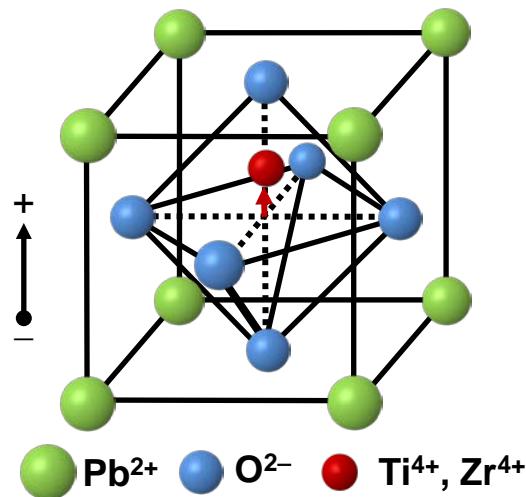
- + Linear dichroism: difference in x-ray absorption for different polarization direction relative to crystalline and/or spin axis.
- + Linear dichroism is due to the anisotropic charge distribution about the absorbing atom caused by bonding and/or magnetic order.
- + “Search Light Effect”:  
X ray absorption of linear polarized x rays proportional to density of empty valence states in direction of electric field vector E.



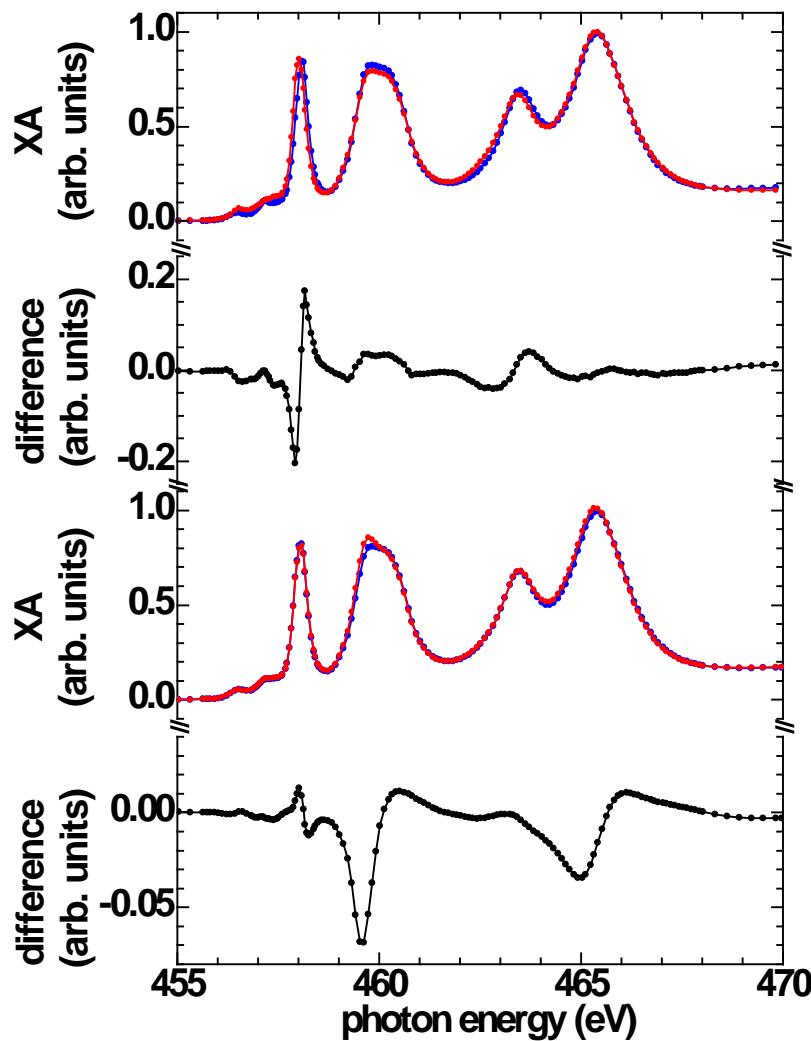
C. T. Chen et al.  
PRL 68, 2543 (1992)



# STRUCTURAL CHANGES IN $\text{PbZr}_{0.2}\text{Ti}_{0.8}\text{O}_3$



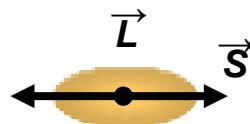
- + Spontaneous electric polarization due to off-center shift of  $\text{Ti}^{4+}, \text{Zr}^{4+}$  ; associated with tetragonal distortion  $\Leftrightarrow$  linear dichroism
- + Reversing the polarization changes XA  $\Leftrightarrow$  Change in tetragonal distortion



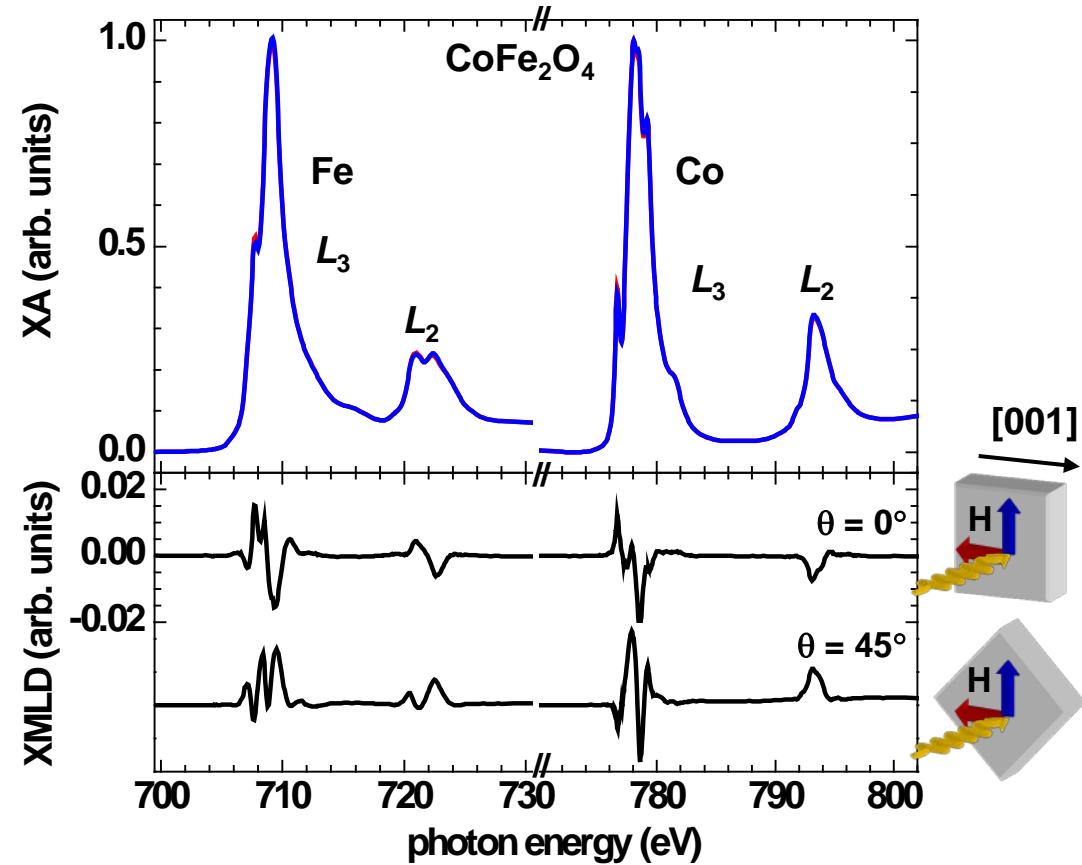
# X-RAY MAGNETIC LINEAR DICHROISM



Isotropic  $d$  electron charge density  
⇒ No polarization dependence

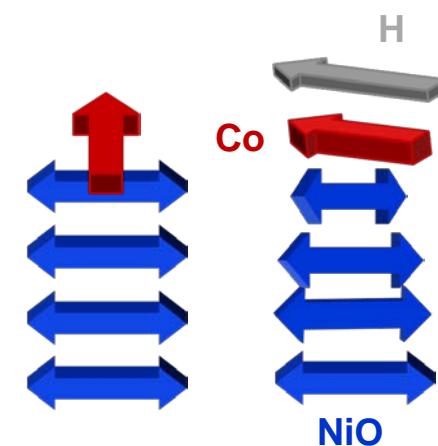
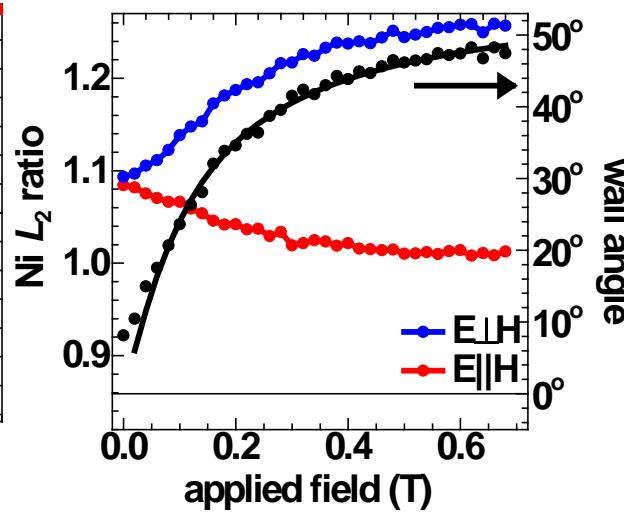
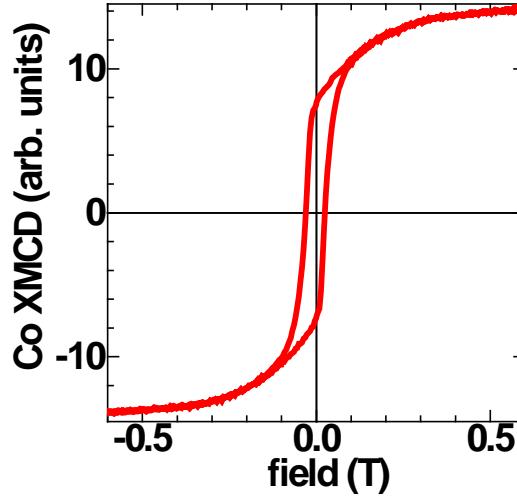
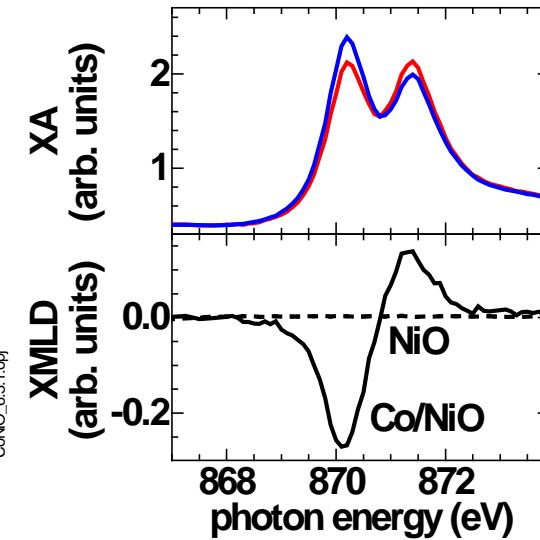
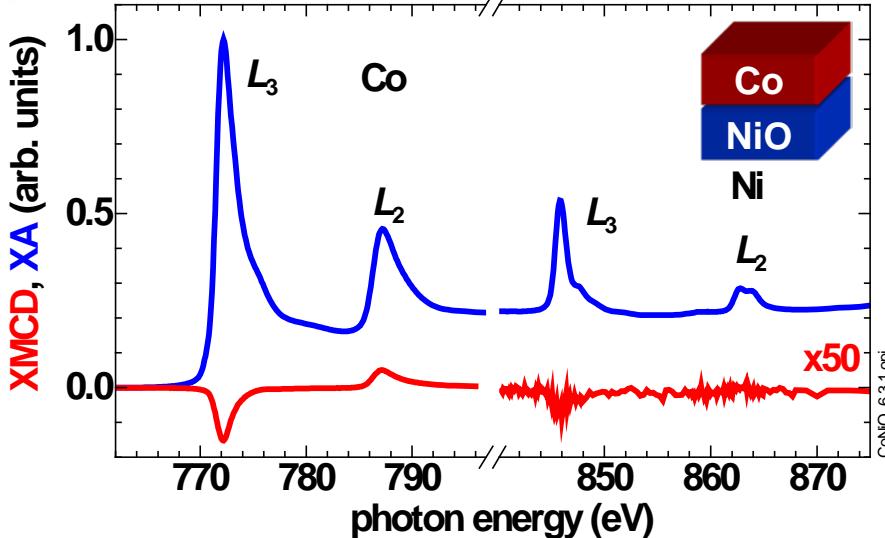


magnetically aligned  
Spin-orbit coupling distorts  
charge density  
⇒ polarization dependence



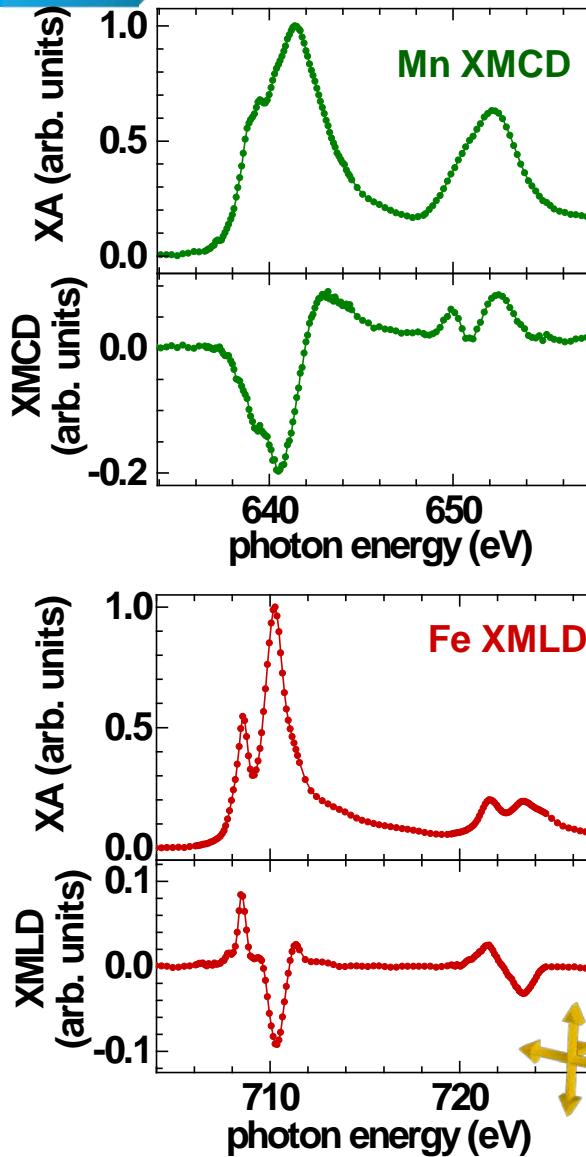
- +  $I_{\text{XMLD}} = I_{||} - I_{\perp} \propto \langle m^2 \rangle$ ,  $\langle m^2 \rangle$  = expectation value of the square of the atomic magnetic moment
- + XMLD allows investigating ferri- and ferromagnets as well as antiferromagnets.
- + XMLD spectral shape and angular dependence are determined by magnetic order and lattice symmetry.

# PLANAR DOMAIN WALL NEAR Co/NiO INTERFACES



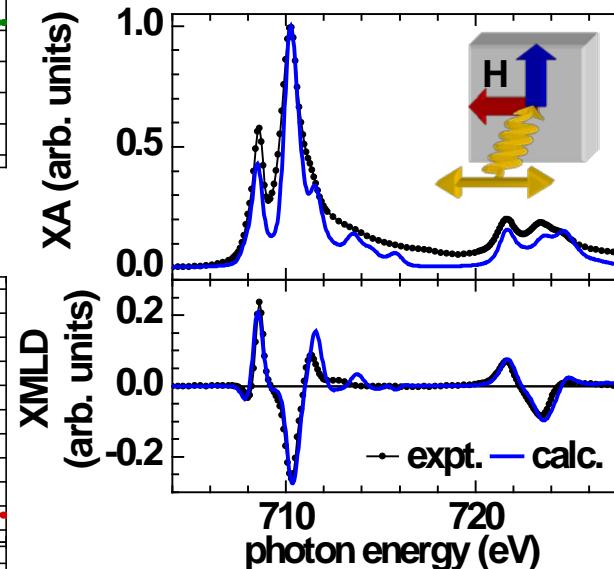
A. Scholl *et al.*,  
Phys. Rev. Lett. **92**, 247201 (2004)

# MAGNETIC COUPLING AT $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3/\text{La}_{0.7}\text{Sr}_{0.3}\text{FeO}_3$ INTERFACES



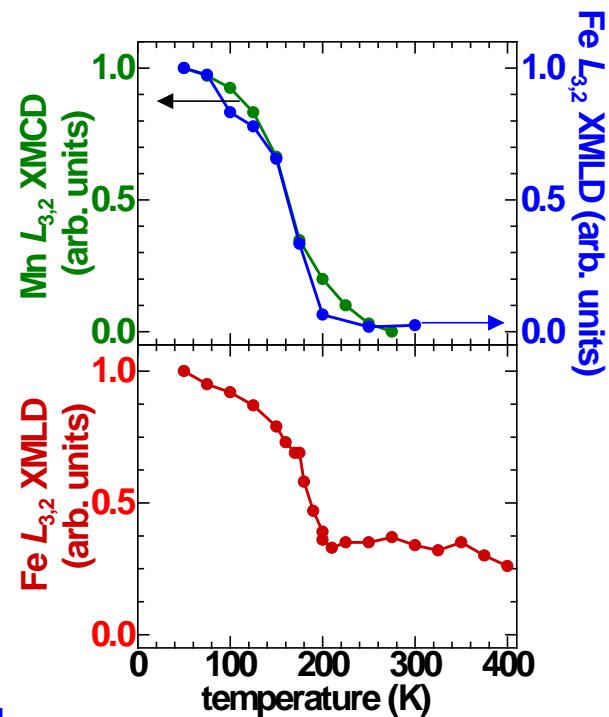
$\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$  (LSMO)  
ferromagnet

$\text{La}_{0.7}\text{Sr}_{0.3}\text{FeO}_3$  (LSFO)  
antiferromagnet

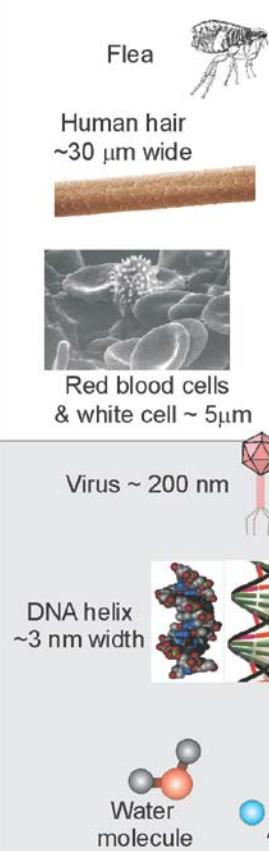


⇒ Perpendicular coupling  
at LSMO/LSFO interface

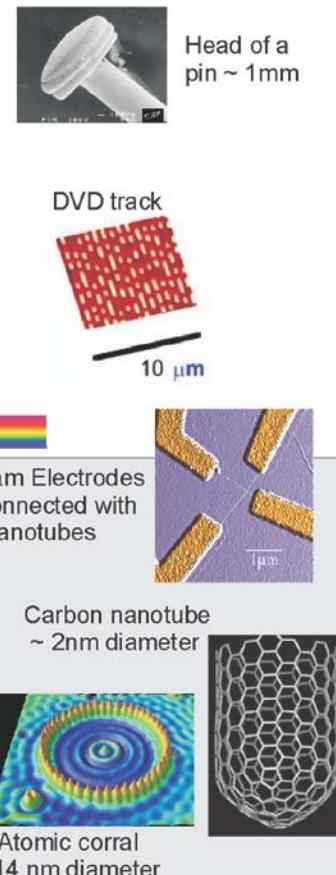
33



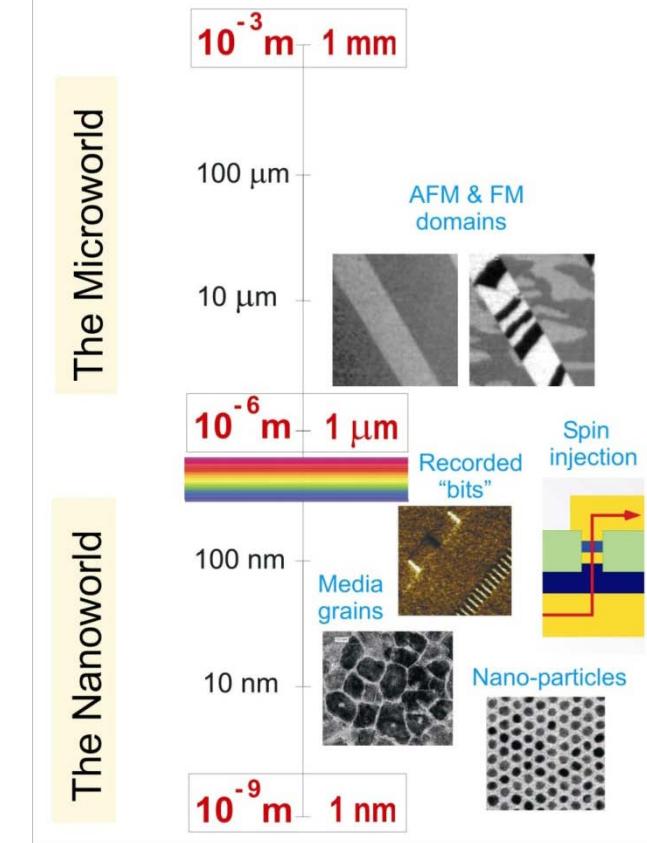
## Nature



## Technology



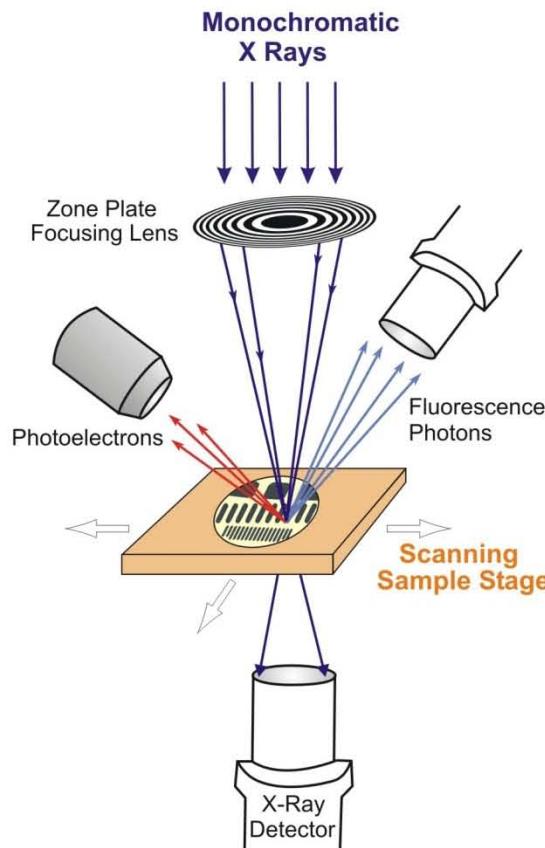
## Magnetism



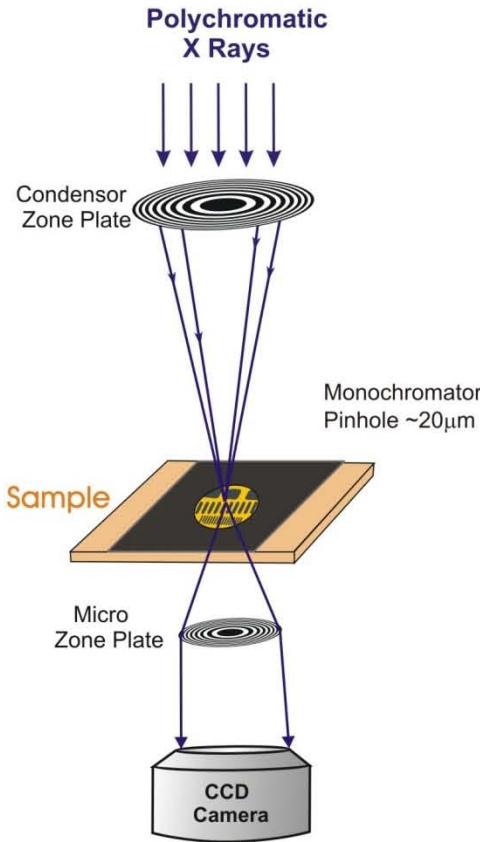
J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

# MAGNETIC MICROSCOPY

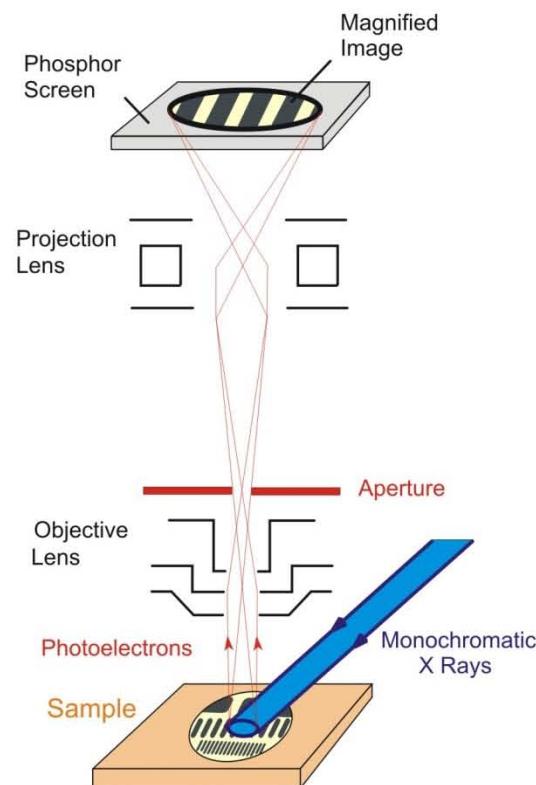
## Scanning Transmission X-ray Microscopy STXM



## Transmission X-ray Microscopy TXM



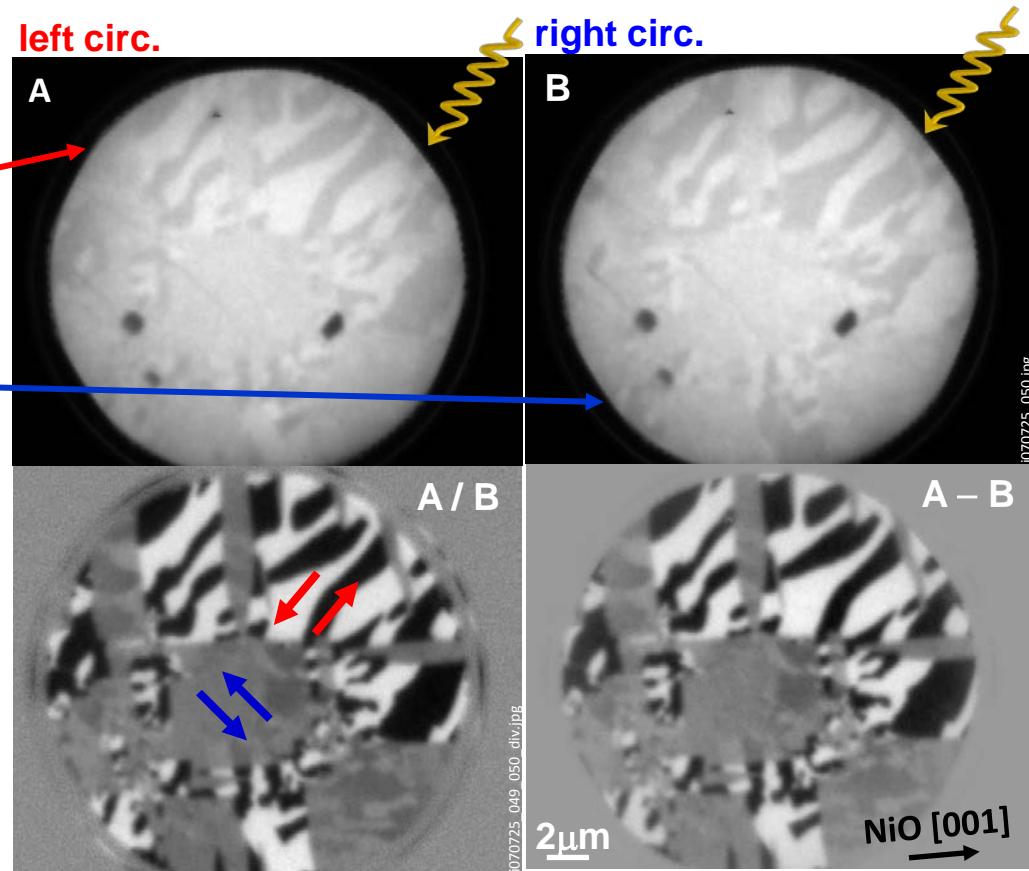
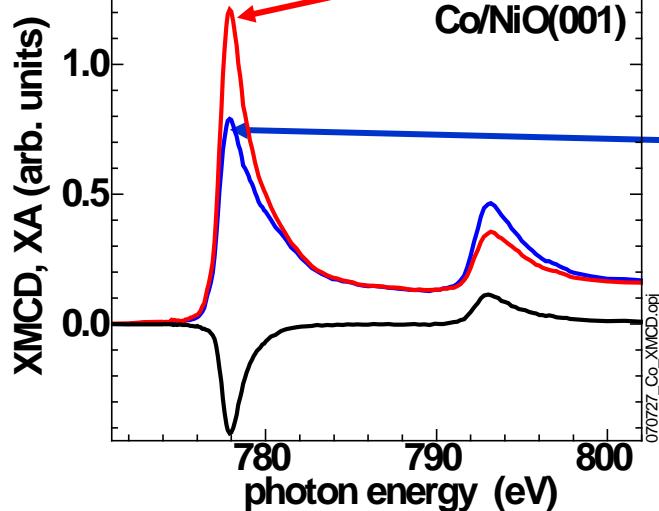
## X-Ray Photoemission Electron Microscopy XPEEM



**10-50 nm spatial resolution**

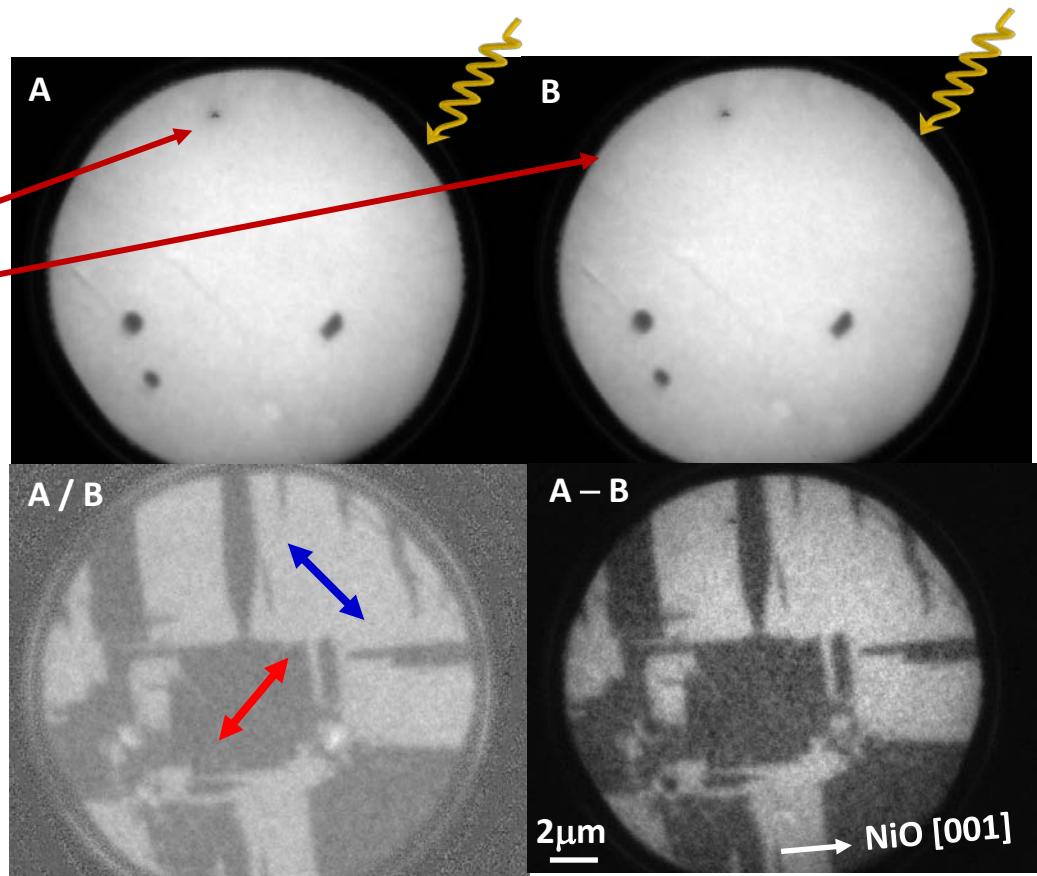
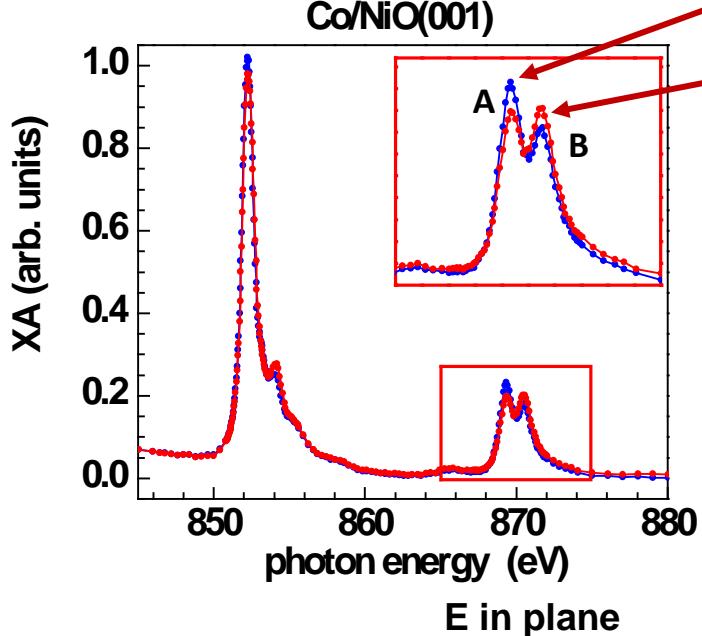
J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

# IMAGING FERROMAGNETIC DOMAINS USING XMCD



- Images taken with left and right circularly polarized x-rays at photon energies with XMCD, i.e. Co  $L_3$  edge, provide magnetic contrast and domain images.

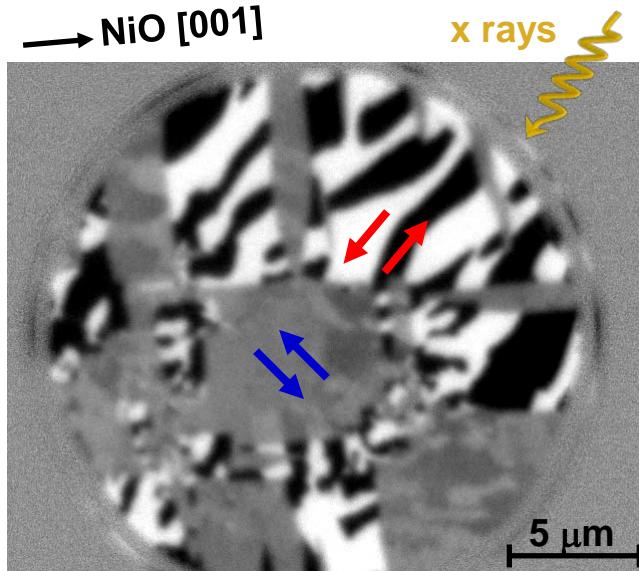
# IMAGING ANTIKERROMAGNETIC DOMAINS USING XMLD



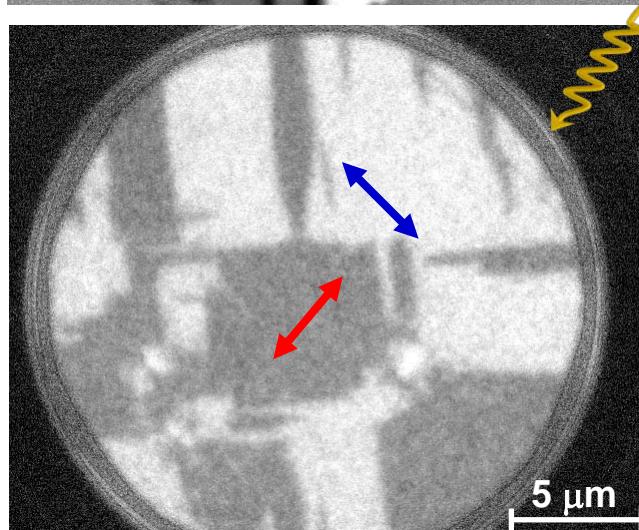
- Images taken with linearly polarized x-rays at photon energies with XMLD, i.e. Ni  $L_2$  edge, provide magnetic contrast and domain images.

# MAGNETIC COUPLING AT Co/NiO INTERFACE

Co XMCD

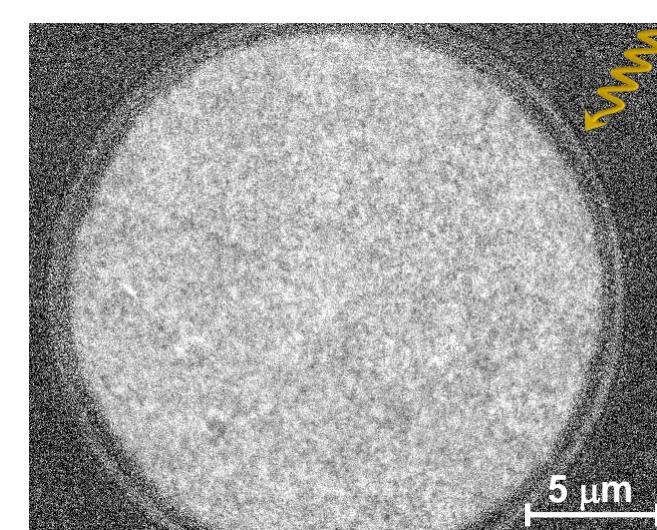


Ni XMLD



probing in-plane

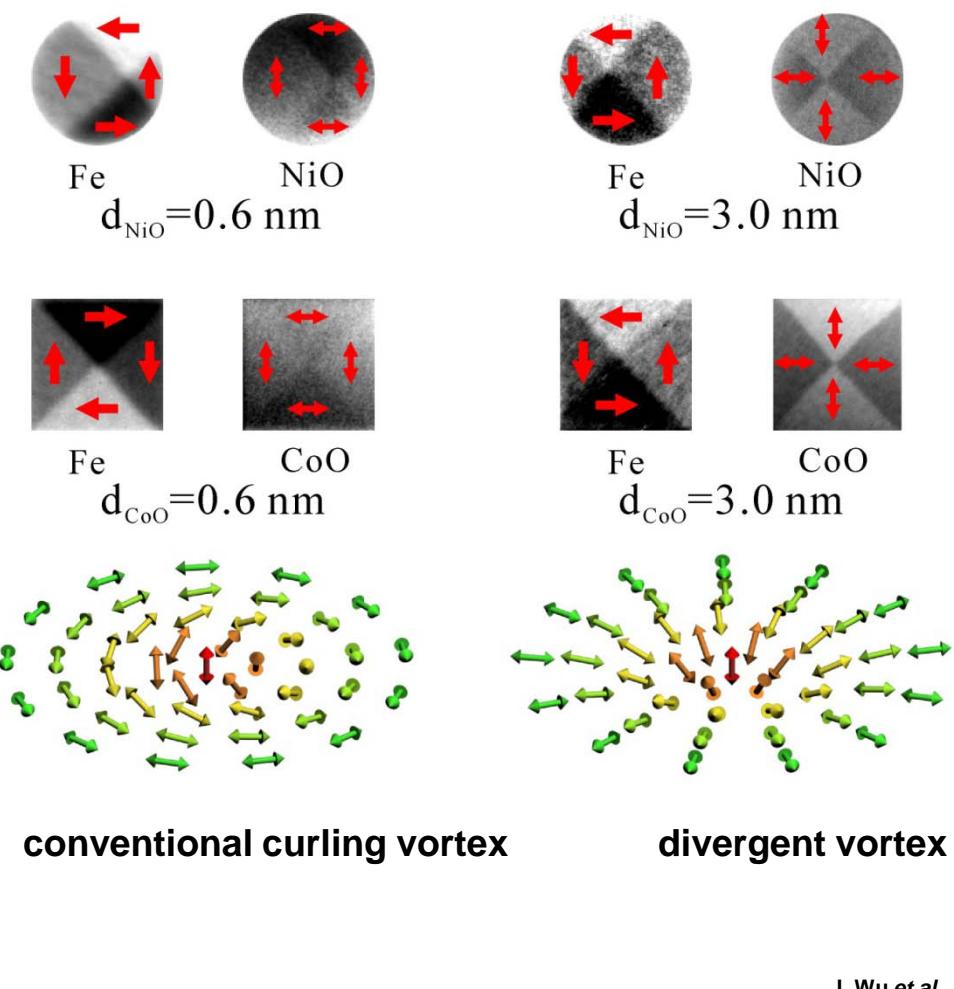
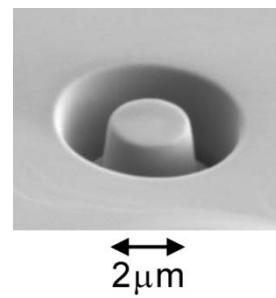
- + Taking into account the geometry dependence of the Ni XMLD signal  
⇒ perpendicular coupling of Co and NiO moments at the interface.



probing out-of-plane

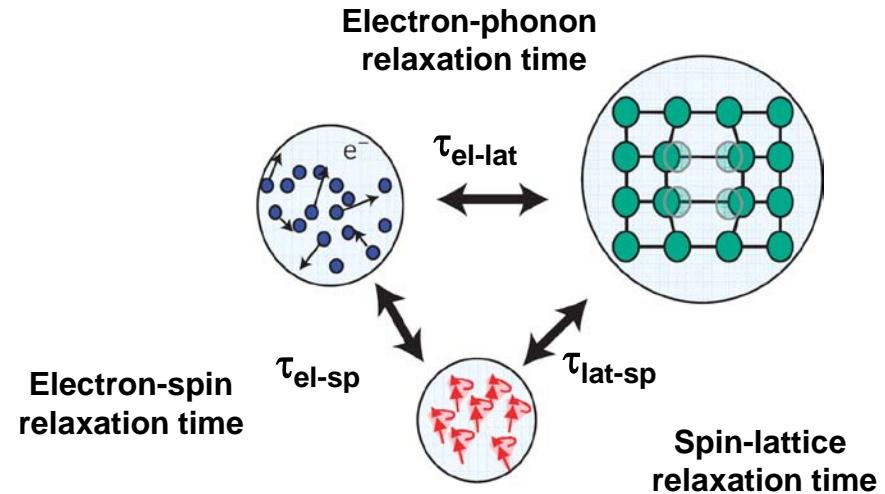
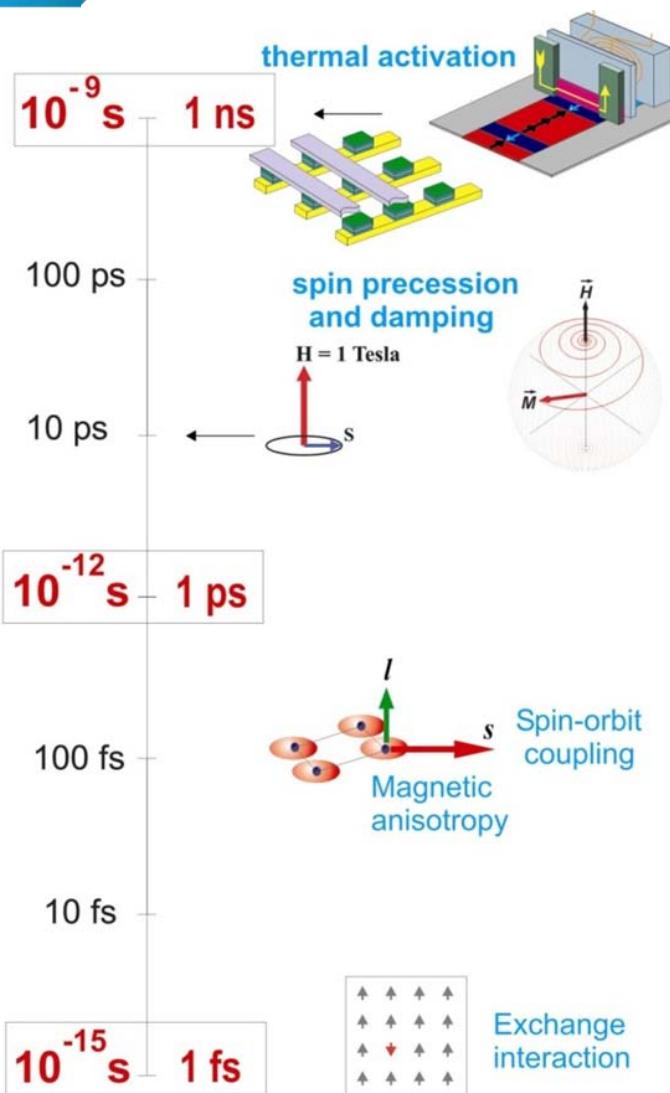
# OBSERVATION OF ANTIFERROMAGNETIC VORTICES

- + First direct observation of vortex state in antiferromagnetic CoO and NiO disks in Fe/CoO and Fe/NiO bilayers using XMCD and XMLD.
- + Two types of AFM vortices:
  - conventional curling vortex as in ferromagnets
  - divergent vortex, forbidden in ferromagnets
  - thickness dependence of magnetic interface coupling



J. Wu et al.,  
Nature Phys. 7, 303 (2011)

# ULTRAFAST MAGNETISM

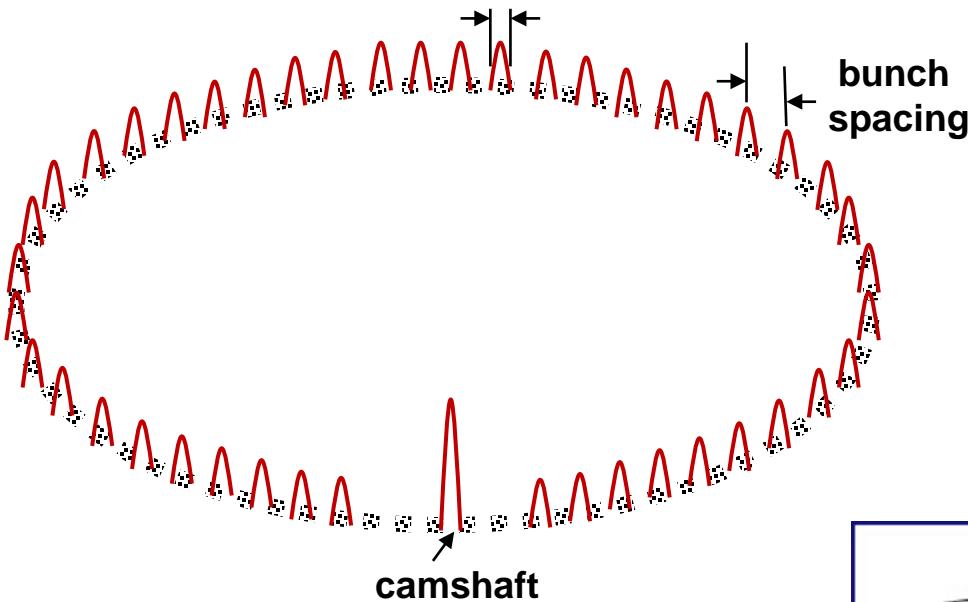


- ✚ Energy reservoirs in a ferromagnetic metal
- ✚ Deposition of energy in one reservoir
- ⇒ Non-equilibrium distribution and subsequent relation through energy and angular momentum exchange

J. Stöhr, H.C. Siegmann,  
Magnetism (Springer)

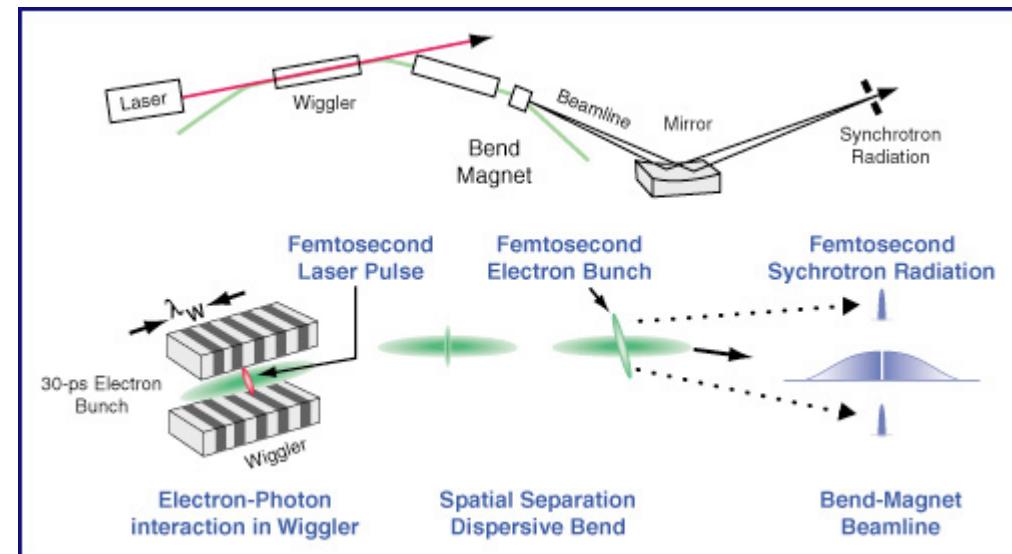
# ALS TIME STRUCTURE

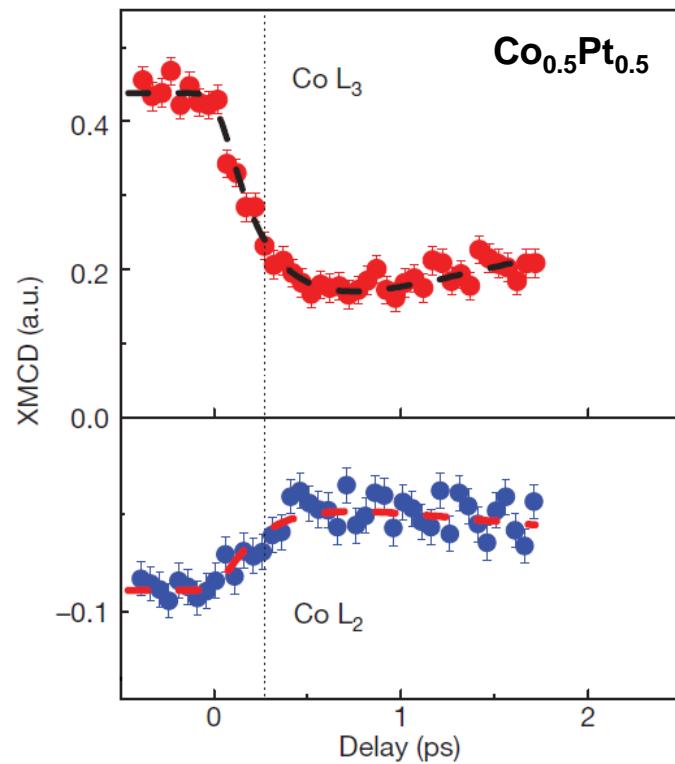
Pulse length 70 ps



- + 256-320 bunches for 500mA beam current
- + Possibility of one or two 5mA "camshaft" bunches in filling gaps
- + Bunch spacing:
  - multibunch mode: 2 ns
  - two-bunch mode: 328 ns
- + Pulse length 70ps

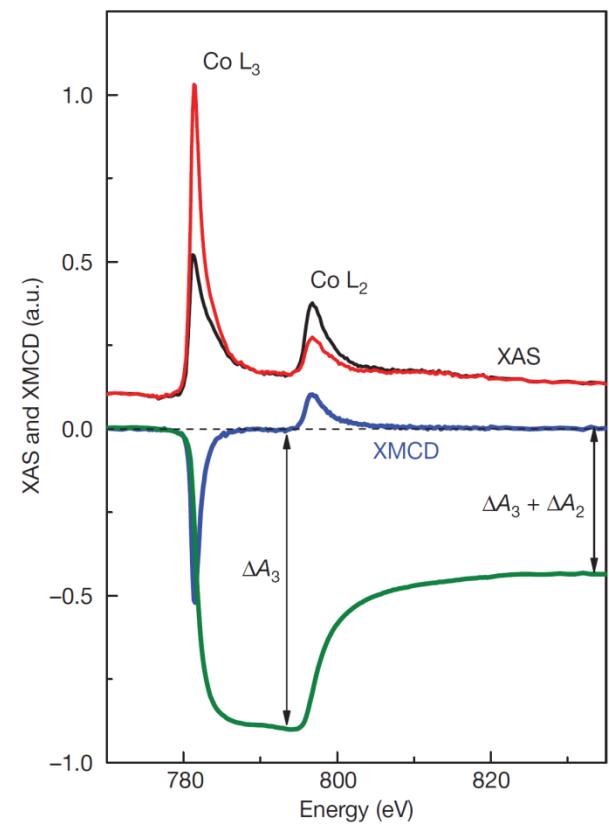
- + <300 fs x ray pulses through "laser bunch-slicing technique"

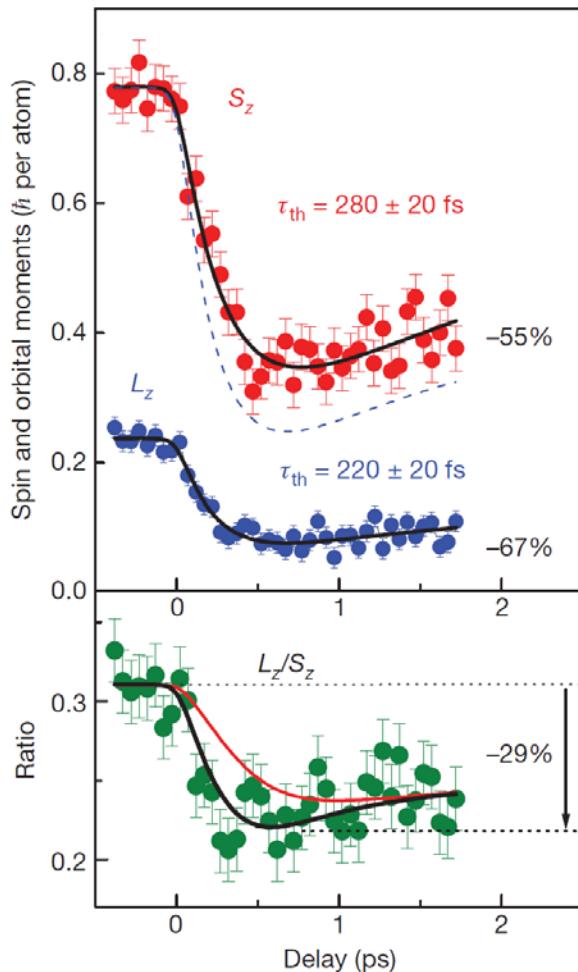




C. Boeglin, et al.,  
Nature 465, 458 (2010)

- + Orbital ( $L$ ) and spin ( $S$ ) magnetic moments can change with total angular momentum is conserved.
- + Efficient transfer between  $L$  and  $S$  through spin-orbit interaction in solids
- + Transfer between  $L$  and  $S$  occurs on fs timescales.
  
- + Co<sub>0.5</sub>Pt<sub>0.5</sub> with perpendicular magnetic anisotropy
- + 60 fs optical laser pulses change magnetization
- + Dynamics probed with XMCD using 120fs x-ray pulses
  
- + Linear relation connects Co  $L_3$  and  $L_2$  XMCD with  $L_z$  and  $S_z$  using sum rules

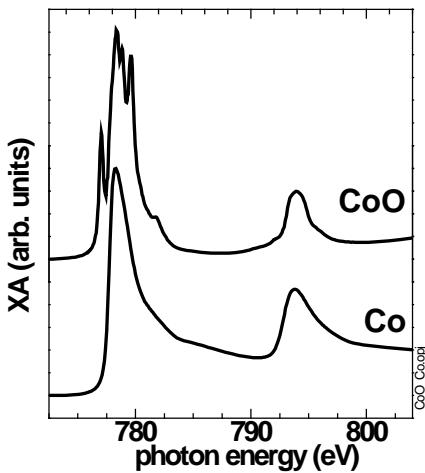




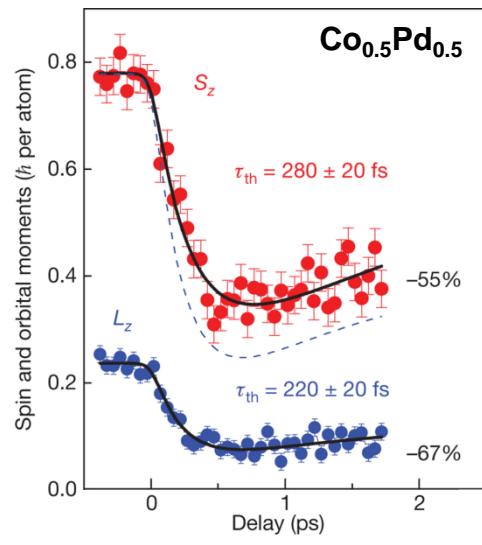
C. Boeglin, et al.,  
Nature 465, 458 (2010)

- + Characteristic thermalization:  
Faster decrease of orbital moment
- + Theory:  
Orbital magnetic moment strongly correlated with magnetocrystalline anisotropy
- + Reduction in orbital moment  
 $\Leftrightarrow$  Reduction in magnetocrystalline anisotropy
- + Typically observed at elevated temperatures in static measurements as well
- + Further studies needed

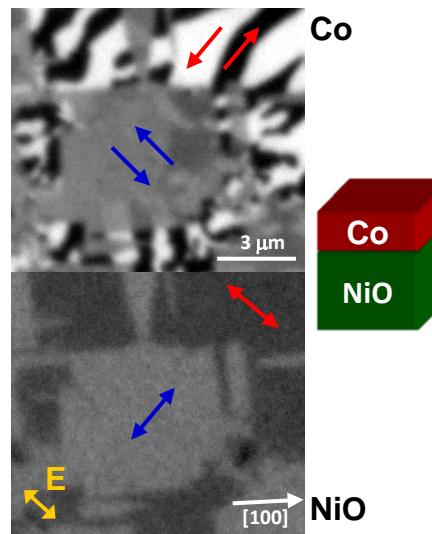
# MAGNETIC SPECTROSCOPY



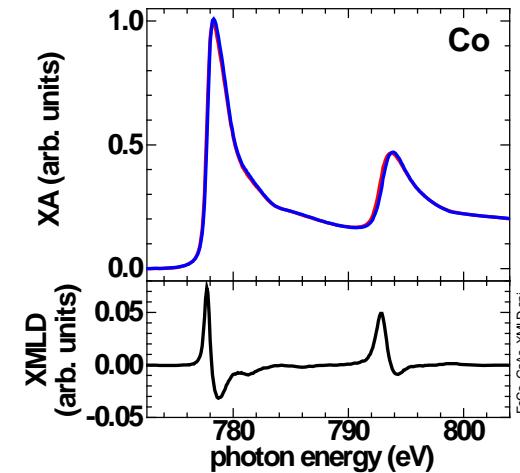
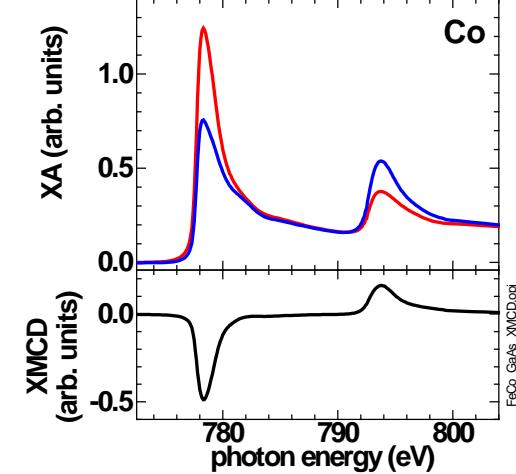
- ✚ X ray absorption
- ✚ X ray magnetic circular dichroism, XMCD
- ✚ X ray magnetic linear dichroism, XMLD
- ✚ X ray magnetic microscopy
- ✚ Magnetization Dynamics



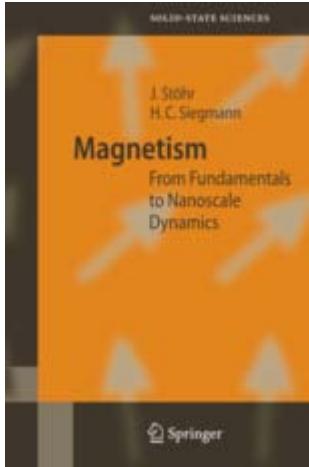
C. Boeglin et al., Nature **465**, 458 (2011)



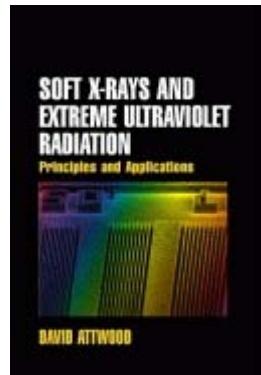
E. Arenholz et al.,  
Appl. Phys. Lett. **93**, 162506 (2008)



## REFERENCES AND FURTHER READING



**J. Stöhr, H.C. Siegmann**  
**Magnetism– From Fundamentals to Nanoscale Dynamics**  
**Springer**



**D. Attwood**  
**Soft X-Rays and Extreme Ultraviolet Radiation:  
Principles and Applications**