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High Curie temperature piezocrystals in the $\text{BiScO}_3\text{-PbTiO}_3$ perovskite system

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Perovskite ferroelectric single crystals in the $(1-x)\text{BiScO}_3\text{-}x\text{PbTiO}_3$ (BSPT) system were grown using a flux method. The Curie temperature of rhombohedral BSPT(43/57) crystals was found to be about 404 °C, with a rhombohedral–tetragonal phase transition temperature (T_{r-t}) around 350 °C. The coercive field (E_c) was found to be 13.7 kV/cm for $\langle 001 \rangle$ -oriented crystals, while the coupling factor k_{33} was determined to be $\sim 90\%$ at room temperature and was nearly temperature independent until 330 °C. High piezoelectric properties together with a high Curie temperature and coercive field make rhombohedral BSPT crystals promising candidates for the next generation of high-performance, high-temperature actuators and transducers. © 2003 American Institute of Physics. [DOI: 10.1063/1.1619207]

For piezoelectric polycrystalline materials, compositions close to a morphotropic phase boundary (MPB) between rhombohedral (R) and tetragonal (T) phases in the solid-solution $\text{PbZrO}_3\text{-PbTiO}_3$ (PZT) are widely utilized in transducer and actuator applications.^{1,2} Since the discovery of PZTs in the 1950s, extensive research has been carried out for the search of new MBP systems. Recently, $\text{Pb}(\text{Zn}_{1/3}\text{Nb}_{2/3})\text{O}_3\text{-PbTiO}_3$ (PZNT) and $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3\text{-PbTiO}_3$ (PMNT) systems have attracted interest in the single-crystal form^{3–7} because of their ultrahigh piezoelectric and electromechanical coupling factors in the $\langle 001 \rangle$ direction, which can reach $d_{33s} > 2000$ pC/N and $k_{33} \sim 94\%$, respectively. This enhanced piezoelectric activity, however, comes with the sacrifice of the temperature stability of the properties, being limited to their relatively low Curie temperatures ($T_C \sim 140\text{--}170$ °C), and further restricted by their lower phase transition temperatures ($T_{r-t} \sim 60\text{--}120$ °C), the consequence of a strong curvature in their MPBs.

In most applications, especially in the automotive and aerospace industries, the need for actuation and sensing over a broad temperature range is essential.⁸ Systems with enhanced piezoelectric activity and higher Curie temperature have redirected effort on single crystals in relaxor-PT systems, including $\text{Pb}(\text{Sc}_{1/2}\text{Nb}_{1/2})\text{O}_3\text{-PbTiO}_3$,⁹ $\text{Pb}(\text{In}_{1/2}\text{Nb}_{1/2})\text{O}_3\text{-PbTiO}_3$,¹⁰ and $\text{Pb}(\text{Yb}_{1/2}\text{Nb}_{1/2})\text{O}_3\text{-PbTiO}_3$ (PYNT).^{11–13} In particular, PYNT offer the highest T_C among all the relaxor-PT systems studied to date with a $T_C \sim 350$ °C. However, as found in PMNT and PZNT single crystals, a strongly curved MPB limits the use of the materials at the $T_{r-t} \sim 160\text{--}170$ °C.¹³ Recently, high-Curie-temperature ferroelectric materials based on $(1-x)\text{BiMeO}_3\text{-}x\text{PbTiO}_3$ solid solutions (where $\text{Me}^{3+} = \text{Sc}, \text{In}, \text{Yb}$, etc.) have been identified.¹⁴ Specifically, $(1-x)\text{BiScO}_3\text{-}x\text{PbTiO}_3$ (BSPT) exhibits a MPB at $x = 0.64$ with a $T_C \sim 450$ °C. Piezoelectric coefficients on the order of 500 pC/N have been demonstrated in polycrystalline ceramics comparable to soft PZTs.¹⁵ In this work, the dielectric and

piezoelectric properties of BSPT single crystals in the R phase close to the MPB were investigated.

BSPT(43/57) single crystals were grown using the high temperature solution method in which Pb_3O_4 and Bi_2O_3 were selected as a self-flux.¹⁶ The as-grown crystals were blue-green in color and rectangular in shape, being 3–8 mm in size. The samples for measurement were oriented along the crystallographic direction $\langle 001 \rangle$ and sputtered with gold. The samples were poled at 120 °C with the application of a 20–30 kV/cm electric field for 10 mins. The room-temperature dielectric permittivity (ϵ_r) of $\langle 001 \rangle$ -oriented BSPT(43/57) crystals was found to be ~ 3000 , with a dielectric loss $\sim 4\%$ after the poling process. Dielectric permittivity and loss as a function of temperature were performed from room temperature to 500 °C using a multifrequency LCR meter (HP4284A) and are shown in Fig. 1. A dielectric maximum was observed at 404 °C, which was assumed to be the Curie temperature, corresponding well to polycrystalline ceramic data.¹⁵ A second dielectric anomaly in Fig. 1 was observed at 349 °C, being the result of a R–T ferroelectric phase transition. It should be noted that the dielectric permit-

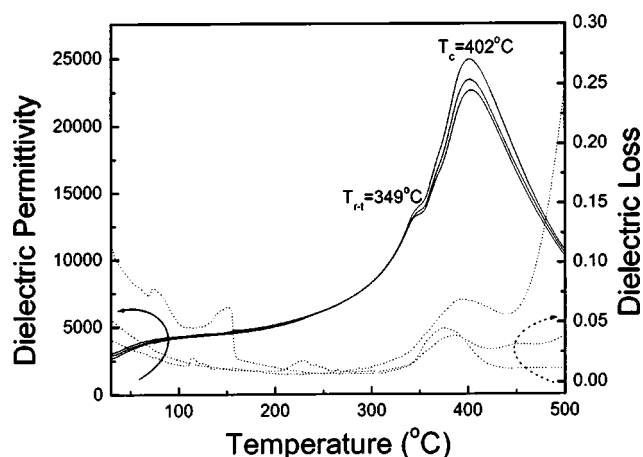


FIG. 1. Dielectric permittivity and loss as a function of temperature for $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals.

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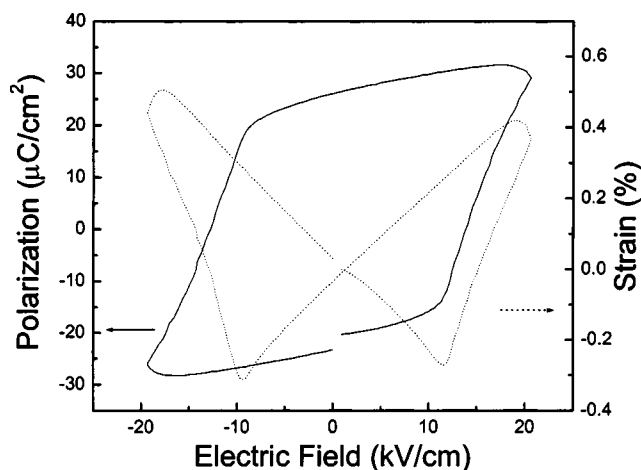


FIG. 2. Bipolar P - E loops and S - E loops for $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals.

tivity is nearly temperature independent in the range of room temperature to 300°C ($\partial\epsilon/\partial T \sim 20/^\circ\text{C}$), indicating thermal stability associated with a high T_C piezoelectrics. The dielectric loss was found to decrease to 2% in the temperature range of 150 to 350°C and reached a peak at the Curie temperature, after which it increased rapidly owing to conductive losses.

Typical hysteresis and strain-field loops determined using a modified Sawyer-Tower circuit driven by a lock-in amplifier (Stanford Research System, Model SR830) at 1 Hz and 20 kV/cm electric fields for $\langle 001 \rangle$ -oriented BSPT(43/57) crystals were measured, and are presented in Fig. 2. The coercive field (E_c) was found to be ~ 13.7 kV/cm with a remnant polarization of $23 \mu\text{C}/\text{cm}^2$. The coercive field is significantly higher than that for PZNT and PMNT systems, being on the order of 2–3 kV/cm.¹² The unipolar strain behavior of $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals exhibited low hysteresis, with strains reaching 0.12% at 10 kV/cm, as shown in Fig. 3. The piezoelectric coefficient d_{33} calculated directly from strain-field curve was ~ 1200 pC/N, similar to that measured by using a Berlincourt d_{33} meter. A maximum strain value of $\sim 1.1\%$ was reached with an electric field of 30 kV/cm, as the crystal transformed from the R to T state with large hysteresis.

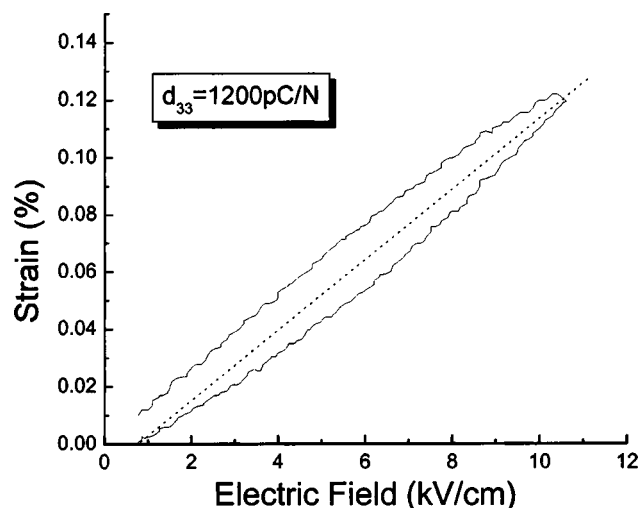


FIG. 3. Unipolar S - E loop for $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals.

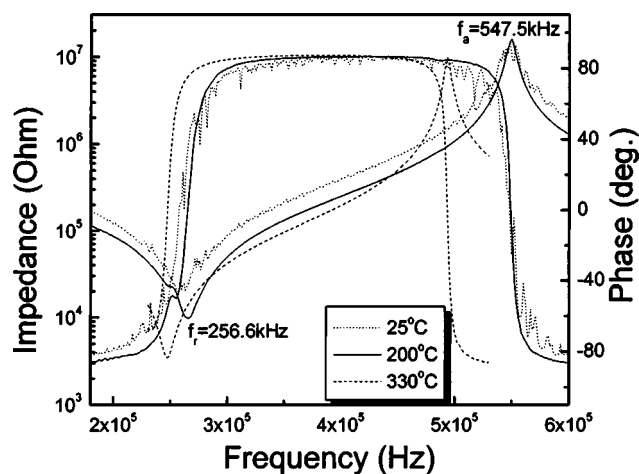


FIG. 4. Resonance and antiresonance frequency characteristics of impedance and phase for the $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals with different temperatures.

The resonance (f_r) and antiresonance (f_a) frequencies for a $\langle 001 \rangle$ -oriented BSPT(43/57) rod (size: $0.6 \times 0.6 \times 3.05 \text{ mm}^3$) measured using an impedance analyzer (HP4194A) were used to determine the electromechanical coupling factor. As shown in Fig. 4, the f_r and f_a were found to be 256.6 and 547.5 kHz at room temperature, respectively, and the longitudinal electromechanical coupling factor k_{33} was calculated to be 90%. The calculated piezoelectric coefficient d_{33} of 1150 pC/N was in good agreement with the values obtained from the unipolar strain measurement. As a function of temperature in the range of 25 to 320°C , both f_r and f_a shifted a little bit with k_{33} decreasing to 89%. Above 330°C , f_a shifted lower and the coupling factor decreased to 88.5%, decreasing further to only 83% when the temperature reached 340°C , as shown in Fig. 5, corresponding to the value of coupling factor for BSPT tetragonal crystal.¹⁶ The same tendencies for the piezoelectric coefficients were also observed and are shown in Fig. 5. This sudden drop above 340°C in the coupling factors and piezoelectric coefficients were the result of the R-T phase transformation as evident in the dielectric behavior.

The various dielectric, piezoelectric, and electromechanical property characterizations of BSPT(43/57) crystals

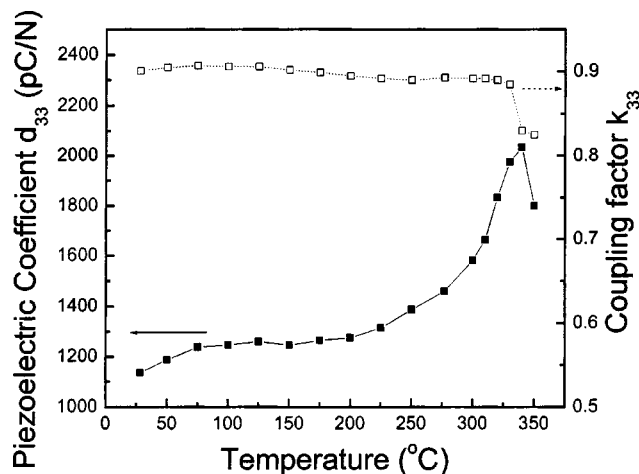


FIG. 5. Longitudinal electromechanical coupling factors k_{33} and piezoelectric coefficients d_{33} as a function of temperature.

TABLE I. Dielectric, piezoelectric, and electromechanical properties of the (001)-oriented BSPT(43/57) single crystals compared with (001)-oriented relaxor-PT single crystals and BSPT ceramics.

	T_C (°C)	T_{r-t} (°C)	K_{33}^T	Loss	P_r ($\mu\text{C}/\text{cm}^2$)	E_c (kV/cm)	d_{33} (pC/N)	k_{33}
BSPT57 crystal ^a	402	349	3000	4%	23	13.7	1150	90%
BSPT64 ceramic ^b	450	/	2010	5%	32	20	460	/
PYNT40 crystal ^c	270	168	2700	2%	26.5	10	1200	88%
PMNT30 crystal	138	98	5000	0.5%	26	2.3	1600	89%
PZNT4.5 crystal	155	120	5000	0.5%	27	3	1800	90%
PZNT8 crystal	170	90	6000	0.5%	30	4	2500	93%

^aThis work.^bSee Ref. 15.^cSee Ref. 17.

are presented in Table I. For comparison, values are also given for various relaxor-PT crystals and polycrystalline BSPT ceramics.

In summary, rhombohedral BSPT(43/57) single crystals possess a high Curie temperature $>400^\circ\text{C}$ and phase transition temperature $>340^\circ\text{C}$, with coercive fields greater than 13 kV/cm. Together with high piezoelectric coefficient $d_{33} \sim 1200$ pC/N, coupling $k_{33} > 90\%$, and temperature-independent properties, BSPT single crystals are attractive candidates for the next-generation high-temperature, high-performance actuators and transducers.

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