

The Cambrian Explosion of Machine Learning Potentials



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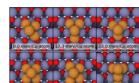
Introduction

RUB

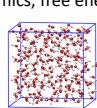
Goal:

Atomic-level understanding of complex systems in chemistry and materials science
⇒ Predictive computer simulations with first-principles quality

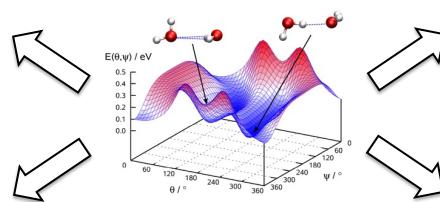
Energy
global and local minima



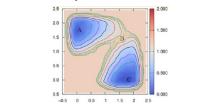
Forces
dynamics, free energies



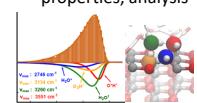
Central Role: Potential Energy Surface



Reactions
barriers / transition states



Vibrations
properties, analysis



⇒ The accuracy of the obtained results depends on the quality of the PES

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Definition of Machine Learning Potentials

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There is no universally accepted definition of machine learning potentials

Challenge: The construction essentially all atomistic potentials involves fitting/parameter optimization.

⇒ Where is the border between fitted potentials and MLPs?

A pragmatic definition of Machine Learning Potentials

- A machine learning potential uses a flexible machine learning method to represent the potential energy surface (total energy and its analytic derivatives) as a function of the atomic coordinates.
⇒ in principle no physical functional form needed
- A machine learning potential is constructed using a consistent set of reference electronic structure calculations.

⇒ *No “artificial intelligence”, just brute force fitting (?)*

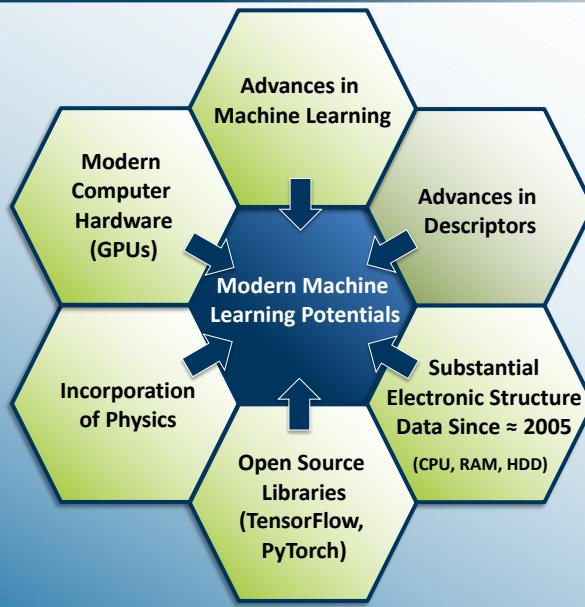
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The Cambrian Explosion of Machine Learning Potentials: Reasons

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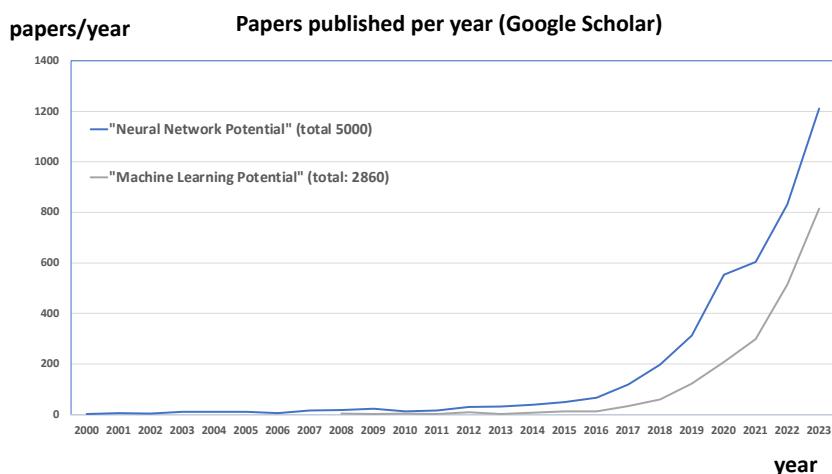
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Machine Learning Potentials

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How to get an overview about the huge amount of papers?

We need a classification scheme for machine learning potentials

What are the criteria?

Historical?

Physics?

Applicability?

Descriptors?

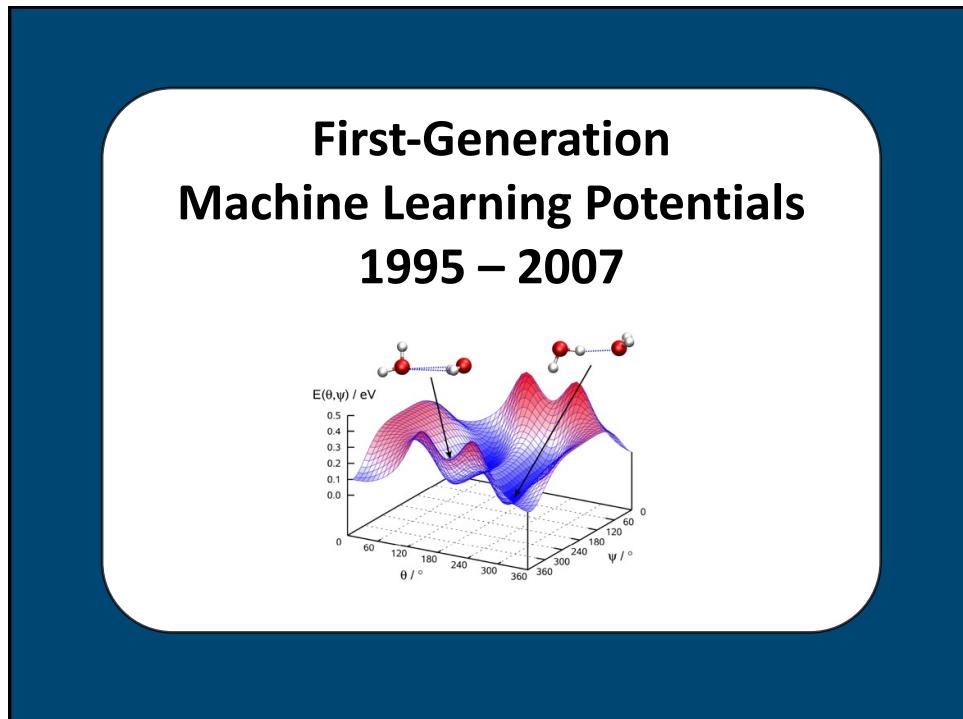
ML Method?

⇒ There is no unique way

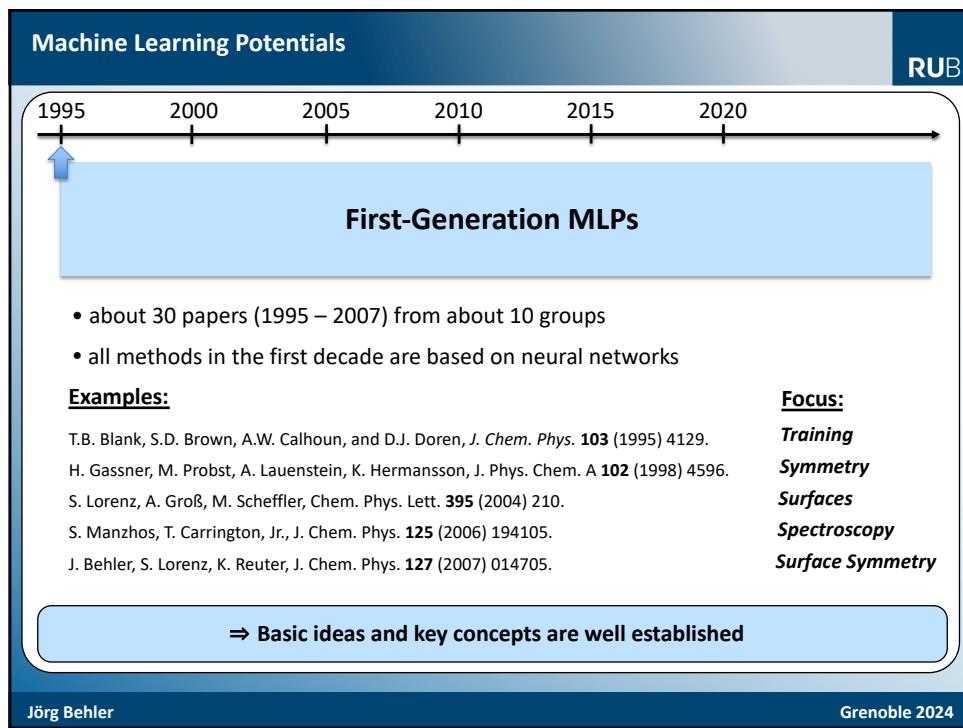
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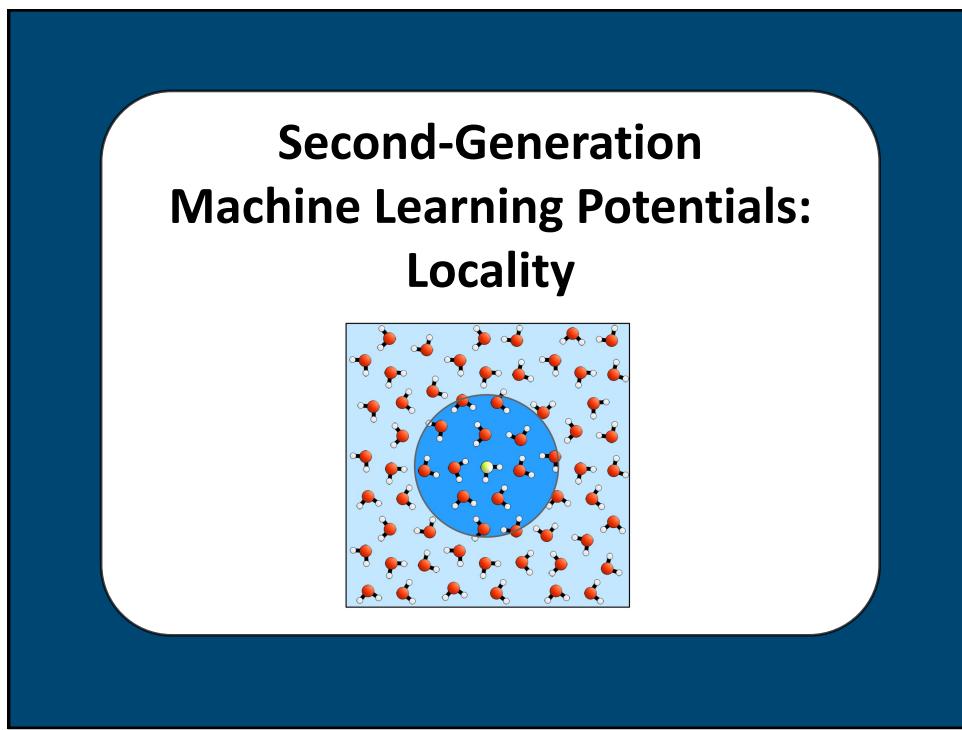
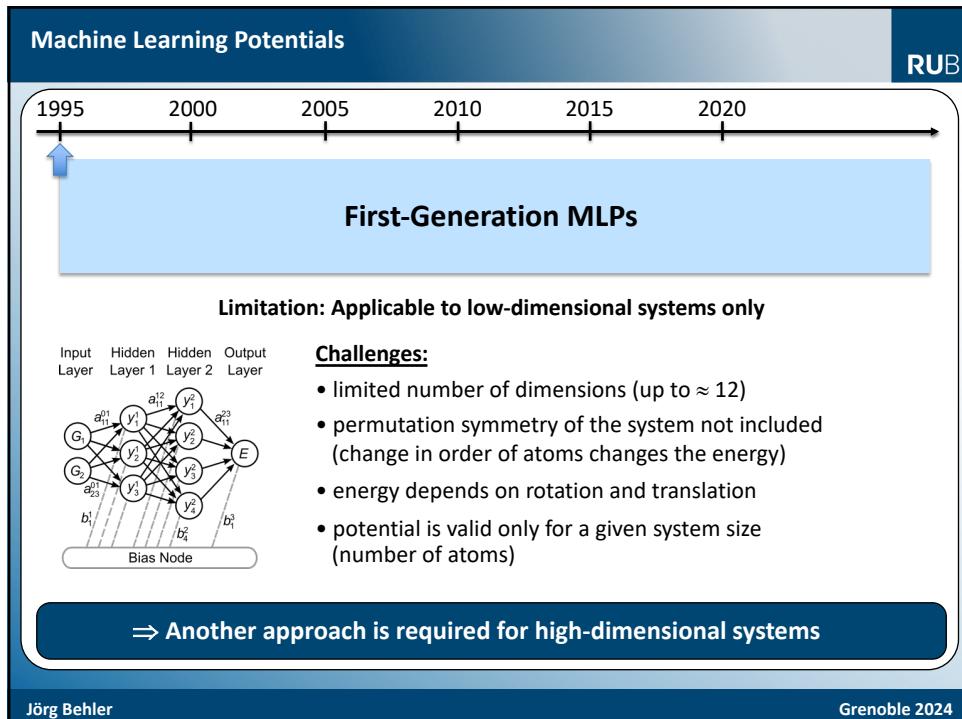
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High-Dimensional Neural Network Potentials

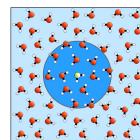
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3 Steps:

1. Total energy is the sum of atomic energies

$$E = \sum_i E_i$$

2. Atomic energies depend on local environments
⇒ cutoff radius



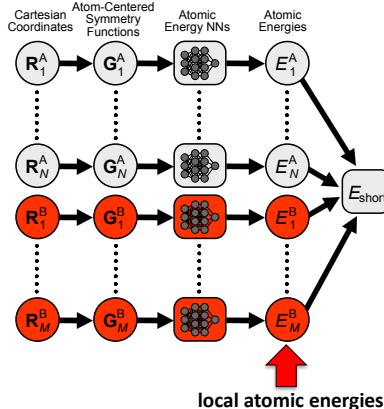
3. Description of local atomic environments by many-body atom-centered symmetry functions
⇒ structural fingerprints
(invariances: rotation, translation, permutation)

J. Behler, M. Parrinello, *Phys. Rev. Lett.* **98** (2007) 146401.

J. Behler, *J. Chem. Phys.* **134** (2011) 074106.

J. Behler, *Angew. Chem. Int. Ed.* **56** (2017) 12828.

Structure for a binary system $A_N B_M$



⇒ applicable to thousands of atoms

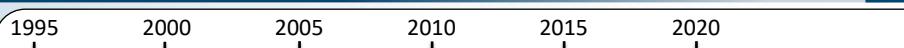
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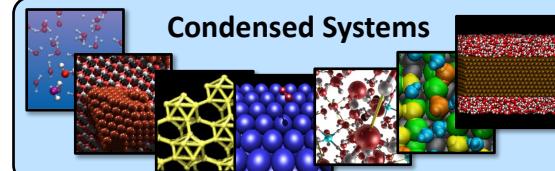
Second-Generation Machine Learning Potentials

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First-Generation MLPs

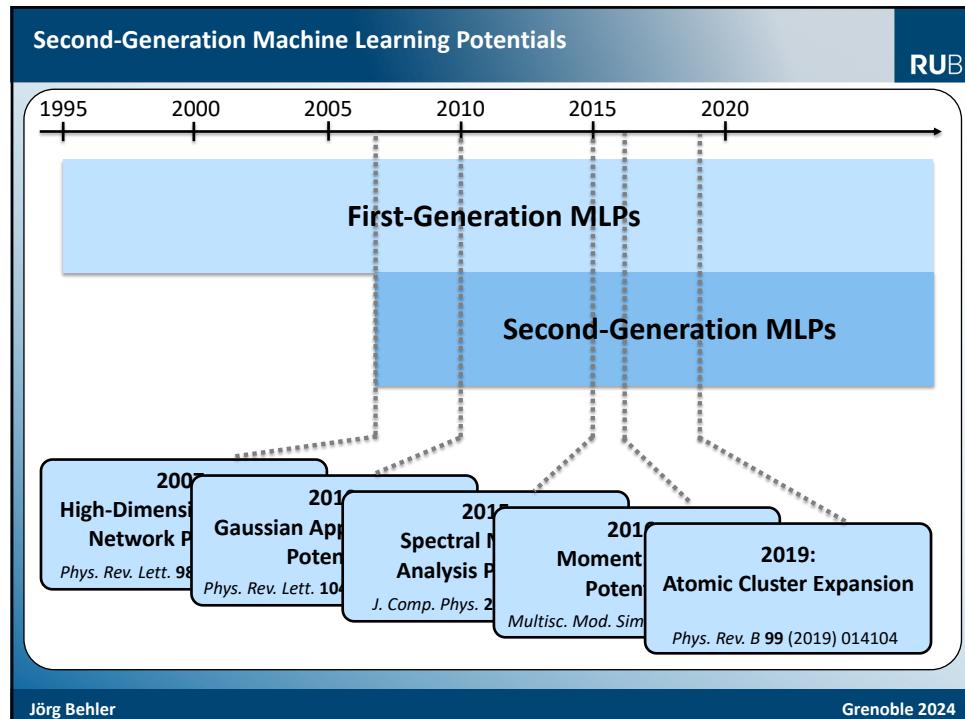
Second-Generation MLPs



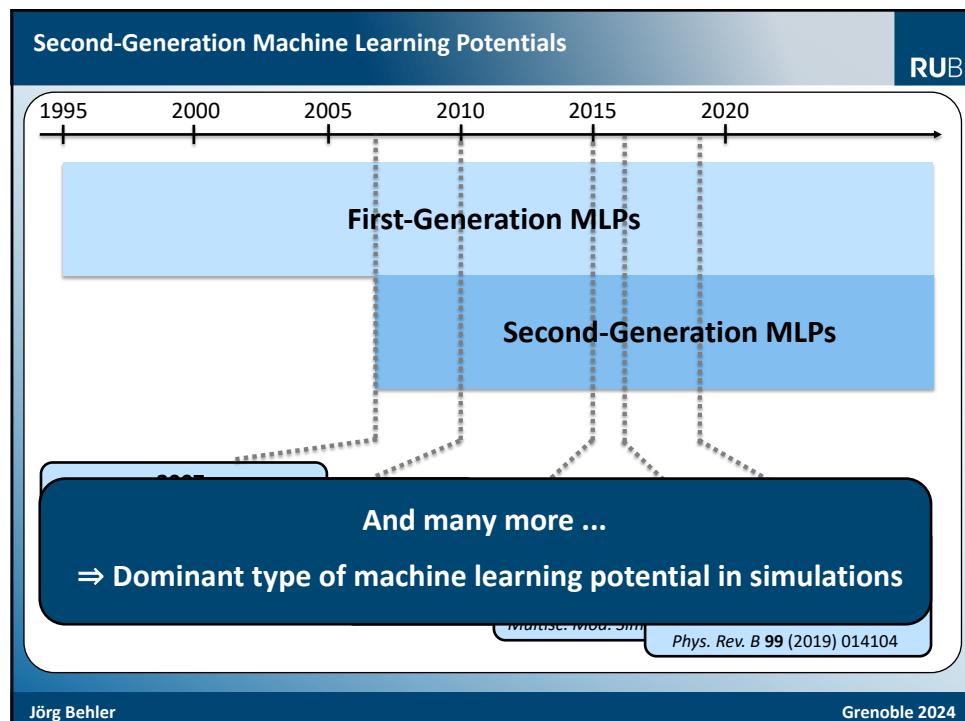
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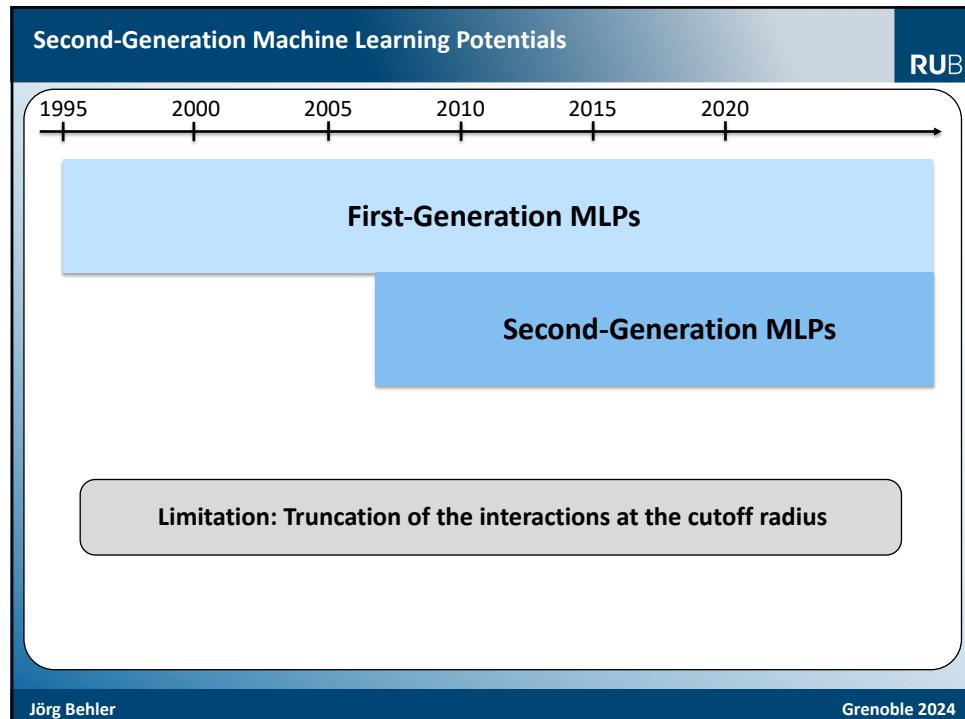
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Hessian-based Locality Test

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A Hessian-Based Analytic Locality Test

Challenge: Atomic forces depend on the entire environment

- different neighbors interact differently
⇒ distance and type of bonding are important
⇒ *influence of individual neighbors* is of interest
- forces are very small or even zero in symmetric environments
⇒ **forces are not a good measure** for interaction strength
- force convergence of central atom measures only total interaction
⇒ **forces can cancel each other** also in non-symmetric environments

Solution:

Hessian: $H_{A_\alpha B_\beta} = \frac{\partial^2 E}{\partial A_\alpha \partial B_\beta} = -\frac{\partial f_{B_\beta}}{\partial A_\alpha} = -\frac{\partial f_{A_\alpha}}{\partial B_\beta}$ with $\alpha, \beta = \{x, y, z\}$

⇒ information about dependence of an atomic force on each atom in the system
⇒ spatial cutoff can be quantified!

M. Herbold and J. Behler, J. Chem. Phys. **156** (2022) 114106.

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Hessian-based Locality Test

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$$\text{DFT Hessian: } H_{A_\alpha B_\beta} = \frac{\partial^2 E}{\partial A_\alpha \partial B_\beta} = -\frac{\partial f_{B_\beta}}{\partial A_\alpha} = -\frac{\partial f_{A_\alpha}}{\partial B_\beta} \quad \text{with } \alpha, \beta = \{x, y, z\}$$

Hessian: (3N x 3N) matrix

	B	1	2	3	4
A	α	x y z	x y z	x y z	x y z
1	x				
2	y				
3	z				
4					

Dependence of atomic force vector on atomic position of a neighbor: (3x3) matrix

⇒ inconvenient, scalar property needed
⇒ Hessian submatrix norm

$$\|\mathbf{h}_{AB}\| = \sqrt{\sum_{\alpha=x,y,z} \sum_{\beta=x,y,z} h_{A_\alpha B_\beta}^2}$$

Dependence of force on each individual atom in the system can be quantified

M. Herbold and J. Behler, J. Chem. Phys. **156** (2022) 114106.

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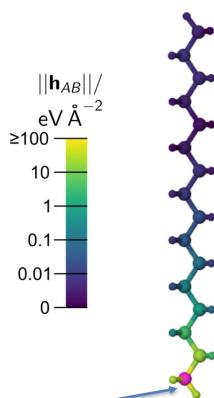
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Hessian-based Locality Test

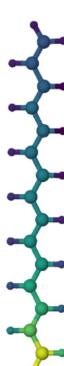
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1D Model Systems

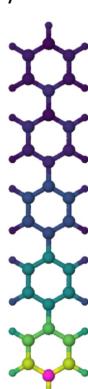
Alkane



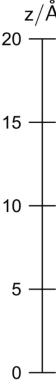
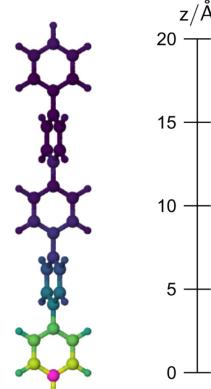
Alkene



Polyaromatic



Localized aromatic



- contribution of each atom in the system can be quantified
- no cancellation of contributions possible

M. Herbold and J. Behler, J. Chem. Phys. **156** (2022) 114106.

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2G MLPs are the main production methods for MLP-driven simulations

There are now two classes:

**Predefined Descriptors
(HDNNP, SOAP, DeePMD, ...)**

- simple \Rightarrow fast
- most applications so far
- more difficult for many elements

**Message Passing Neural Networks
(DTNN, SchNet, MACE, ...)**

- iterative extension of environment
- learnable features
 \Rightarrow more demanding

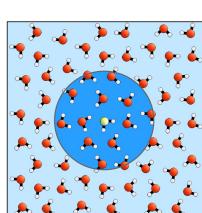
\Rightarrow for a given system the accuracy is very similar

If the training of the MLP for a system does not work, there is a physical reason
 \Rightarrow must be understood

- 1) Data is not accurate enough
- 2) Descriptors are insufficient
 \Rightarrow properties of different structures are averaged \Rightarrow large errors
- 3) Physics is missing

The cutoff used in second-generation MLPs is an approximation!

\Rightarrow Convergence tests are needed



Locality tests:

Statistical analysis:

V. L. Deringer and G. Csányi, Phys. Rev. B **95** (2017) 094203.

Hessian-based:

M. Herbold and J. Behler, J. Chem. Phys. **156** (2022) 114106.

\Rightarrow For some systems, long-range interactions may be important!

Third-Generation Machine Learning Potentials: Long-Range Interactions

What about electrostatics?

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Learning Electrostatic Multipoles

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Goal:

Use ML to represent flexible electrostatic charges and multipoles as a function of the molecular structure.

⇒ improved description of electrostatics in classical force fields for small molecules

Main methods: Kriging and Neural Networks

⇒ not a full MLP, since only the electrostatic component is constructed using ML

S. Houlding, S. Y. Liem and P. L. A. Popelier, Int. J. Quant. Chem. **107** (2007) 2817.

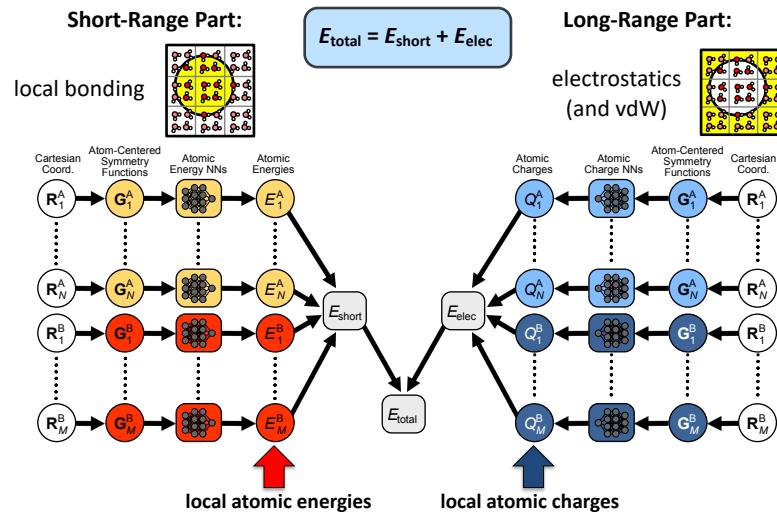
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Third-Generation Neural Network Potentials: Method

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N. Artrith, T. Morawietz and J. Behler, *Phys. Rev. B* **83** (2011) 153101.
T. Morawietz and J. Behler, *J. Chem. Phys.* **136** (2012) 064103.

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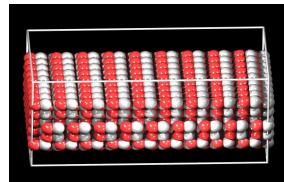
Machine Learning Potentials

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1995 2000 2005 2010 2015 2020

First-generation MLPs

Zinc Oxide Surface



Second-generation MLPs

Third-generation MLPs

Examples:

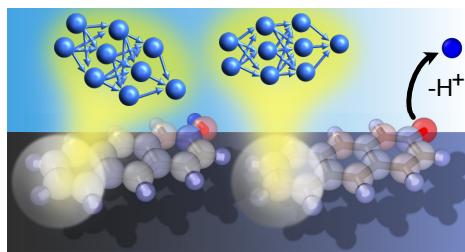
- N. Artrith, T. Morawietz and J. Behler, *Phys. Rev. B* **83** (2011) 153101.
- T. Morawietz and J. Behler, *J. Chem. Phys.* **136** (2012) 064103.
- K. Yao et al., *Chem. Sci.* **9** (2018) 2261.
- O. T. Unke and M. Meuwly, *J. Chem. Theory Comput.* **15** (2019) 3678.

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Fourth-Generation Machine Learning Potentials: Global Charge Distribution



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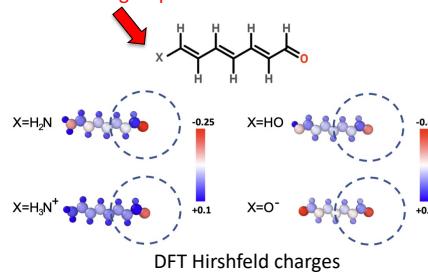
Limitations of Local Machine Learning Potentials

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Effect of distance structural changes in molecular systems:

Example 1: C₇H₇O_X

functional group



⇒ different local properties for the same local geometry
⇒ contradictory training data

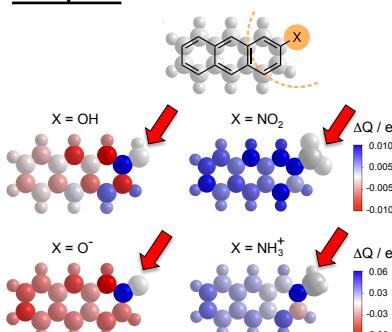
Example 2: anthracene

X = OH

X = NO₂

X = O⁻

X = NH₃⁺



⇒ charge transfer determines reactivity
(e.g. mesomeric effect)

T. W. Ko, J. A. Finkler, S. Goedecker, J. Behler, Nature Commun. 12 (2021) 398.

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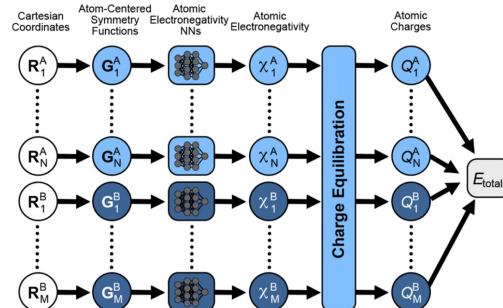
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CENT – Charge Equilibration Neural Network Technique

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$$E_{\text{tot}}(\{Q_i\}) = \sum_{i=1}^{N_{\text{atom}}} \left(E_i^0 + \chi_i Q_i + \frac{1}{2} J_{ii} Q_i^2 \right) + \frac{1}{2} \iint \frac{\rho(\mathbf{R}) \rho(\mathbf{R}')}{|\mathbf{R} - \mathbf{R}'|} d\mathbf{R} d\mathbf{R}'$$

electronegativities hardness Coulomb energy



⇒ global electronic structure included (non-local charge transfer)
 ⇒ applications: systems with primarily ionic bonding

S. A. Ghasemi, A. Hofstetter, S. Saha, S. Goedecker, Phys. Rev. B **92** (2015) 045131.

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Fourth-Generation Neural Network Potentials: Method

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Goal: Combination of the advantages of CENT and HDNNPs

CENT
global charge distribution
⇒ electrostatics

2G-HDNNP
atomic energies
⇒ local bonding

modified training:
atomic charge

additional atomic descriptor:
atomic charge

4G-HDNNP
 $E_{\text{total}} = E_{\text{short}} + E_{\text{elec}}$
atomic energies + global charges
⇒ local bonding + electrostatics

T. W. Ko, J. A. Finkler, S. Goedecker, J. Behler, Nature Commun. **12** (2021) 398.

T. W. Ko, J. A. Finkler, S. Goedecker, J. Behler, Acc. Chem. Res. **54** (2021) 808.

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Fourth-Generation Neural Network Potentials: Method

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Goal: Combination of the advantages of CENT and HDNNPs

glob
⇒ e

criptor:

Advantages:

- applicable to all types of bonding and systems (ionic, covalent, metallic, ...)
- long-range electrostatic interactions (flexible charges)
- description of non-local charge transfer
- applicable to multiple global charge states

atomic energies + global charges
⇒ local bonding + electrostatics

T. W. Ko, J. A. Finkler, S. Goedecker, J. Behler, Nature Commun. 12 (2021) 398.

T. W. Ko, J. A. Finkler, S. Goedecker, J. Behler, Acc. Chem. Res. 54 (2021) 808.

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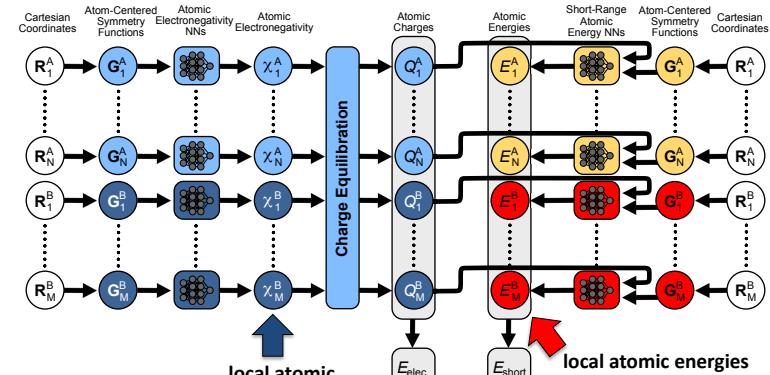
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Fourth-Generation Neural Network Potentials: Method

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Non-Local Long-Range Part

Short-Range Part



$$E_{total} = E_{short} + E_{elec}$$

→ global electronic structure included (non-local charge transfer)

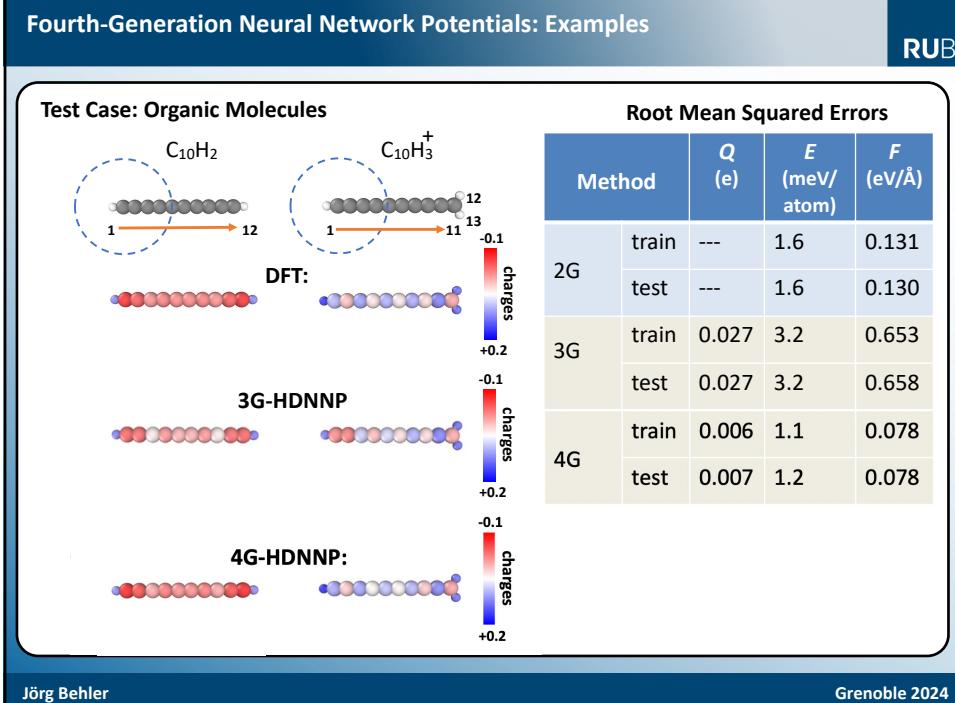
T. W. Ko, J. A. Finkler, S. Goedecker, J. Behler, Nature Commun. 12 (2021) 398.

T. W. Ko, J. A. Finkler, S. Goedecker, J. Behler, Acc. Chem. Res. 54 (2021) 808.

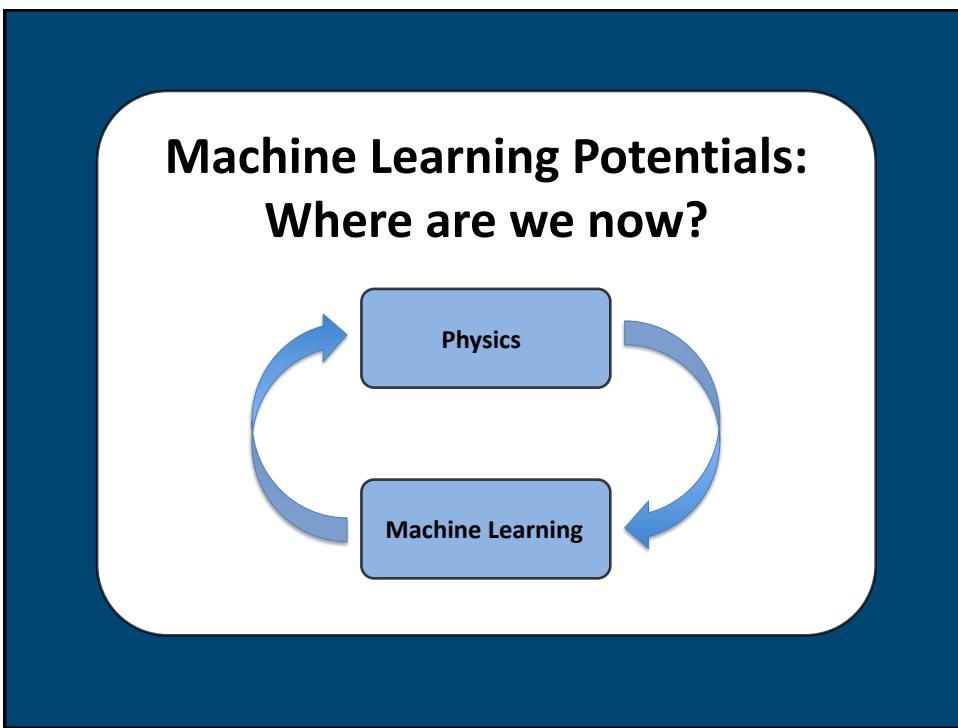
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Challenge: How to measure the accuracy?

To date strong focus on E and F RMSEs: "the lower the better..."

But we need to know:

- accuracy for unknown (relevant) structures
- stability in simulations
- are the trajectories correct

We need to check the right properties

⇒ a lot of effort, requires physical knowledge about the system

What is our benchmark? What is the truth?

- Theory? Experiment?

Coupled cluster calculations for large systems are impossible!

⇒ Often we do not know the truth

Good news: In DFT we have accepted this dilemma and it works!

What to do?

- **The MLP must have a low RMSE for E and F**

Mandatory but not a sufficient criterion

⇒ large remaining errors indicate problems (data, descriptor, physics...)

- **Molecular dynamics must be long-term stable**

But: long-term stability is no proof for the right physics ("*Seeing is believing*"?)

"*Well, you can run it...*"

- **Understand the physics of the system before training a potential**

⇒ check the critical properties

Comparison to experiment is often difficult

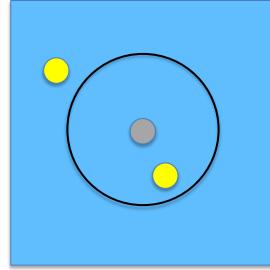
Validation of MLPs is become a main task

Example: Multiple Charge States

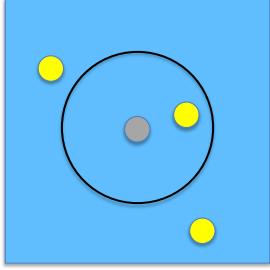
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The $\text{Fe}^{2+}/\text{Fe}^{3+}$ redox system in water RUB

FeCl_2 in water



FeCl_3 in water



Expectation:

- A local second-generation MLP must fail for this combined system
- A global fourth-generation MLP should work for this system

How can we investigate/validate this?

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The Fe²⁺/Fe³⁺ redox system in water

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Construction of 6 HDNNPs

HDNNP	Dataset	E RMSE (meV/atom)	F RMSE (eV/Å)	Q RMSE (me)
2G(Fe ²⁺)	FeCl ₂ /water	0.203 (0.213)	0.032 (0.032)	---
2G(Fe ³⁺)	FeCl ₃ /water	0.232 (0.245)	0.037 (0.037)	---
2G(Fe ²⁺ /Fe ³⁺)	all	0.262 (0.271)	0.034 (0.035)	---
4G(Fe ²⁺)	FeCl ₂ /water	0.203 (0.258)	0.031 (0.036)	2.896 (2.896)
4G(Fe ³⁺)	FeCl ₃ /water	0.237 (0.258)	0.036 (0.036)	3.130 (3.124)
4G(Fe ²⁺ /Fe ³⁺)	all	0.258 (0.273)	0.033 (0.033)	3.105 (3.108)

⇒ no significant difference, all very accurate (?)

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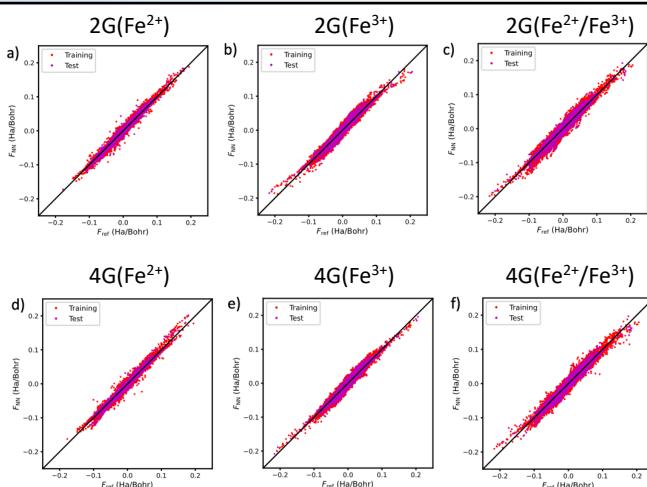
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The Fe²⁺/Fe³⁺ redox system in water

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What about outliers? Force correlation plots



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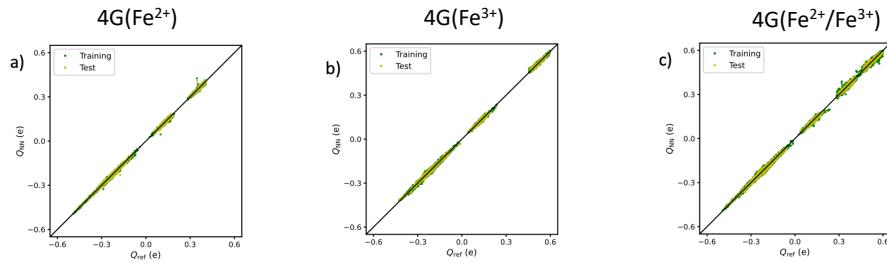
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The Fe²⁺/Fe³⁺ redox system in water

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4G-HDNNP Charge Correlation



⇒ Charges of both oxidation states are well represented

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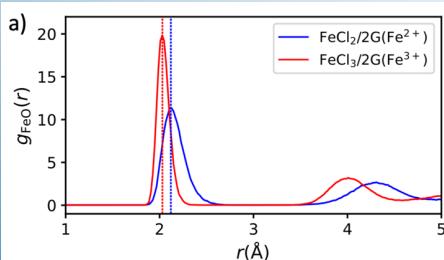
The Fe²⁺/Fe³⁺ redox system in water

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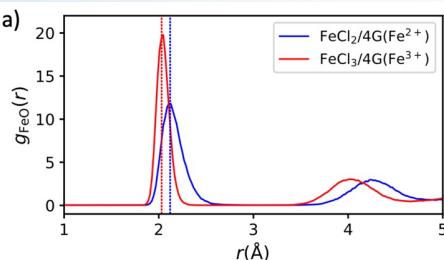
Fe-O Radial Distribution Functions:

Test 1: HDNNPs trained only to FeCl₂ OR FeCl₃ (15 Å box)

2G-HDNNP



4G-HDNNP



⇒ All trajectories are stable

⇒ For separate datasets all HDNNPs work fine

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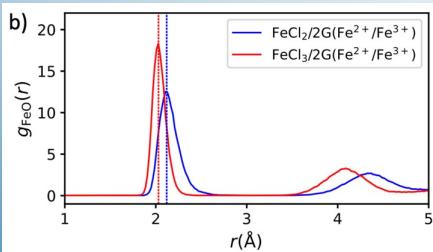
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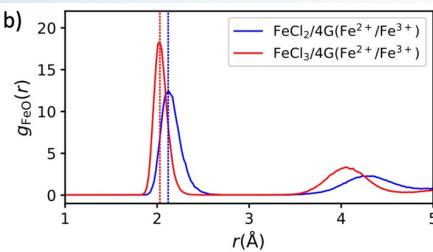
Fe-O Radial Distribution Functions:

Test 2: HDNNPs trained to FeCl₂ AND FeCl₃ (15 Å box)

2G-HDNNP



4G-HDNNP



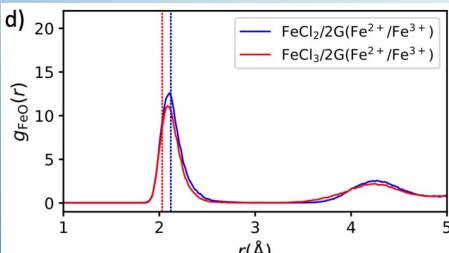
⇒ All trajectories are stable

⇒ For combined datasets all HDNNPs *SEEM* to work fine

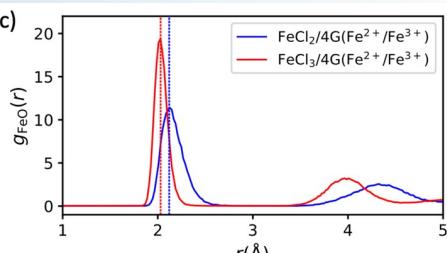
Fe-O Radial Distribution Functions:

Test 3: HDNNPs trained to FeCl₂ AND FeCl₃ (30 Å box)

2G-HDNNP



4G-HDNNP



⇒ 2G-HDNNPs fail, 4G-HDNNPs are correct

The Fe²⁺/Fe³⁺ redox system in water

RUB

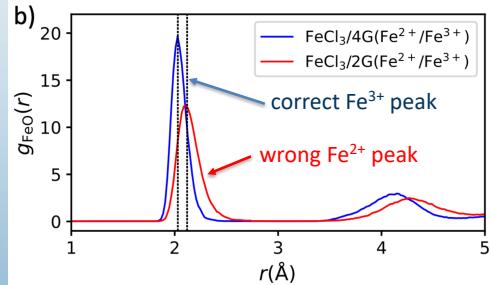
Fe-O Radial Distribution Functions:

Test 4: HDNNPs trained to FeCl₂ AND FeCl₃ – Addition of Cl atom

Initial structure:
FeCl₂ in water



Newly equilibrated system:



⇒ 2G-HDNNPs fail, 4G-HDNNPs are correct

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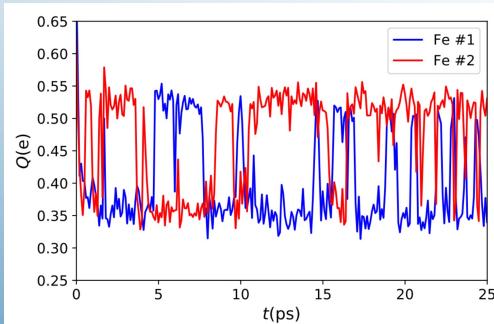
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Extrapolation Test: Electron Transfer Reaction

Test: Fe₂Cl₅ in a large box of water



⇒ 4G-HDNNP can predict electron transfer and charge conservation

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Conclusions

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Conclusions		RUB
Physics remains important	⇒ Right model for the system	
Real applications are important	⇒ We must go beyond model toy systems	
ML not good for extrapolation	⇒ Range of validity is essential information	
Reference methods are important	⇒ MLP error is smaller than DFT error	
Reporting RMSEs is not enough	⇒ Stability in simulations is essential	
⇒ Using Machine Learning Potentials requires physical understanding and validation		
⇒ There is still a lot of room for further methodical developments		

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