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HA2: Shape of aluminium nanoparticles

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Task N ^o	Points	Avail. points
Σ		

Introduction

The properties of nanoparticles differ immensely from the equivalents in bulk materials. However, the properties are not only dependent on the size of the particles; their shape turns out to play an important role as well. Usually the shape can experimentally be controlled by the adjustment of various parameters during the synthesis. In that way non-equilibrated (from a thermodynamic point of view) particles can be made by e.g. applying stabilizing organic molecules onto the surface. However, to predict the thermodynamic shape (corresponding to the one with the lowest energy for a given number of particles) the so-called Wulff construction may be used. In this construction the surface energies of the nanoparticle's various facets are essential[1]. This study computationally investigates such Wulff constructions for large aluminium nanoparticles by applying density functional theory (DFT), with a restriction to the (111) and the (100) facets. The goal is to calculate the most stable shape of these nanoparticles and examine how external conditions (such as molecules in gas phase) affect the shape.

Computational method

DFT is in principle exact. However, the exact form of the exchange-correlation potential is not known and hence various approximations are made to solve specific problems. The approximations can become increasingly intricate and elaborate, often with accompanying increase in the computational complexity and cost. A commonly used approximation is to treat the exchange-correlation potential as a second degree Taylor potential in the electron density. !! This violates properties of the XC potential which are known to be true. How and why can we use this then!!

There are many different GGA functionals and no single one is optimal for all applications. For metals, the PBE, or simply the LSDA functional will capture the relevant physical properties of the system as the density does not vary rapidly [].

Atop of xc-functionals, an appropriate basis set needs to be specified for the problem at hand. Commonly used basis sets are linear combination of atomic orbitals (LCAO), a set of plane waves, and real space grids. A difficulty with LCAO is that one needs sufficient information about the system at hand to make an educated guess about the atomic orbitals. However, the computational process is easy to parallelize and is generally easily computed since the solution for the orbitals can be extrapolated in space. The real space method is generally computationally expensive (however often exact). Plane waves have neat physical properties, such as that the energy is only dependent on the wave-vector and that solutions become more exact when one increases the number of plane waves. A problem with the plane waves basis set is that it utilizes Fourier transforms which are generally tedious to parallelize. We did not investigate the atomic orbitals of aluminum enough to make an educated guess and we tried to avoid unreasonably heavy computational burden. On top of this, we really enjoyed that the plane waves converge to an accurate solution as the basis set expanded. Thus, we decided to use the plane wave basis set in the simulation of the aluminum nanoparticles.

Another thing to consider is the discrete k spacing for the energy bands. The k-space domain needs to be sampled at fine enough distances to capture most parts of the band structure. If k values are sampled too far apart, the energy can differ from the actual energy, both an increase and a decrease. To get a sufficient amount of k-points is often not the most computationally heavy step and therefore can be exaggerated a bit. To find a reasonable value for the spacing, simulations were made with plane wave sets with increasing energy, and the spacing for which the total energy had stabilized was used in further simulations.

To get a good estimate of the surface energy one has to include atoms of the bulk since they effect the interatomic distance near the surface, and hence the surface energy. The surface approaches a stable energy at about XXXX atoms, see figure XXXXX.

The simulation was made both in vacuum and with CO molecules attached to the surface. The CO molecules were attached on top of aluminium atoms in both the 111 and 100 surface. The occupancy of atoms was set to 1/4 by adsorbing one CO molecule to a 2x2 surface. **Is it ok to do it this way?**

When updating the density in the self consistency loop, it is useful to keep some information about the previous density when navigating in the energy landscape. (The

information about the direction towards the minimal energy should not be lost in each iteration step). This is implemented with a so called mixer. The new density is usually updated according to the following rule:

$$n^{new} = \beta n^{old} + (1 - \beta)n^{new} \quad (1)$$

For metals, the recommended value for β is about 0.1. GPAW has a build in mixer, which takes in three arguments, β , number of previous generations used 'nmaxold' and a weight. For the two latter parameters, the default value was used, 5 generations weight 50.

Results

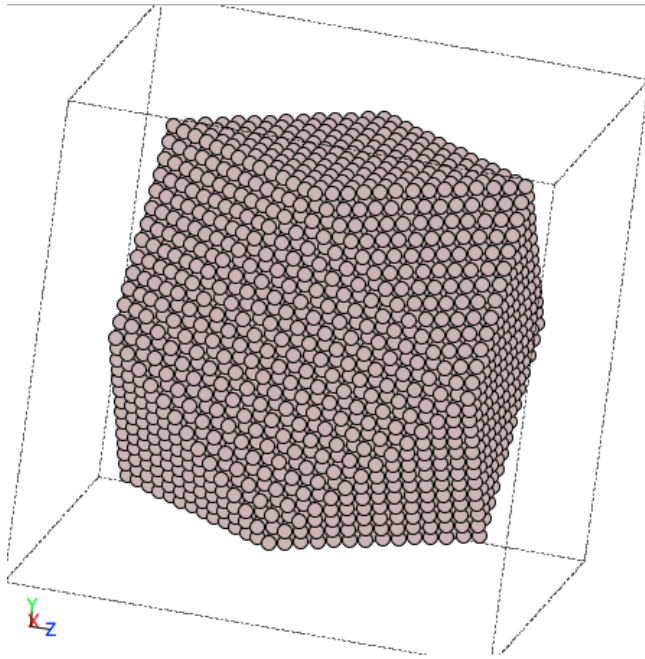


Figure 1: Wulff construction of aluminum in vacuum

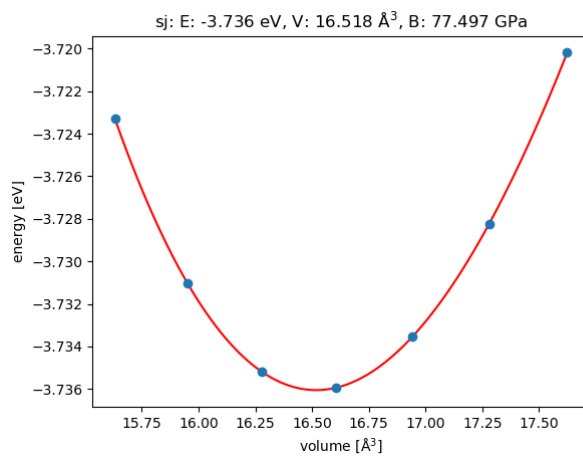


Figure 2: Total lattice energy for varying lattice spacings

Different shapes of wulf construction

Discussion

(Try to bring order in your results, rationalize the results to the best of your ability)

Conclusion

References

- [1] Magnus Rahm, *Task and background*. Pingpong - TIF035 Computational Materials Physics VT18, 2018.
<https://pingpong.chalmers.se/courseId/9094/content.do?id=4260146>. Retrieved 2018-02-02.

A Source Code

A.1 Main simulation: /HA2_simulation.py

```
1  #!/usr/bin/env python
2  # coding=utf-8
3
4  ### imports ###
5  import numpy as np
6  from gpaw import GPAW, Mixer, PW
7  from ase.build import *
8  from ase.io import read
9  from ase.units import kJ, J, m
10 from ase.eos import EquationOfState
11 from ase.build import fcc111, fcc100, add_adsorbate
12 from ase.cluster.wulff import wulff_construction
13 from ase.visualize import view
14 from ase.parallel import rank
15 from ase import Atoms
16
17 ### Constants and parameters ###
18 a_al = 4.05 # Lattice constant for Al, experimentally determined
19 N_lattice_spacings = 7 # Number of lattice constants to loop over to find ←
    equilibrium
20 E_al = 84.67567 # Ionization energy for hardest bound core electron
21 theta = 1 / 4.0 # Monolayer coverage
22
23 ### Create Al bulk and initialize calculator parameters ###
24 al = bulk('Al', 'fcc', a=a_al, cubic=False) # Create Al bulk
25 mixer = Mixer(beta=0.1, nmaxold=5, weight=50.0) # Recommended values for small ←
    systems
26 E_cut = [50, E_al, 100, 200, 300, 400, 500] # cut-off energy
27
28 for energy in [500]: # Change to E_cut to loop and check convergence
29     calc = GPAW(mode=PW(energy), # use the LCAO basis mode
30                 h=0.18, # grid spacing, recommended value in this course
31                 xc='PBE', # Exchange-correlation functional
32                 mixer=mixer,
33                 kpts=(12, 12, 12), # k-point grid - LOOP OVER LATER TO CHECK "←
    CONVERGENCE"
34                 txt='out.txt') # name of GPAW output text file
35     al.set_calculator(calc)
36
37 ### Find lattice constant with lowest energy ###
38 cell_0 = al.cell # Unit cell object of the Al bulk
39 for eps in np.linspace(-0.02, 0.02, N_lattice_spacings):
40     al.cell = (1 + eps) * cell_0 # Adjust lattice constant of unit cell
41     al.get_potential_energy() # Calculate the potential energy for the Al ←
    bulk
42
43 confs = read('out.txt@0:' + str(N_lattice_spacings)) # Read the ←
    c o n figurations
44
45 # Extract volumes and energies:
46 volumes = [atoms.get_volume() for atoms in confs]
47 energies = [atoms.get_potential_energy() for atoms in confs]
48 # if rank == 0:
49 #     print energies, shape(energies)
50
51 # Plot energies as a function of unit cell volume (directly related to latt. ←
    const.)
52 eos = EquationOfState(volumes, energies)
53 v0, E_bulk, B = eos.fit()
54 eos.plot('Al_eos.png')
55 a_calc = (4 * v0)**(1 / 3.0) # Latt. const. acc. to ASE doc., but why is ←
    this correct?
56
57 N_x = 1
58 N_y = 1
59 N_z = 6
60
61 surface111 = fcc111('Al', size=(N_x, N_y, N_z), a=a_calc, vacuum=7.5)
62 surface100 = fcc100('Al', size=(N_x, N_y, N_z), a=a_calc, vacuum=7.5)
63 surface111.center(axis=2)
64 surface100.center(axis=2)
65
66 # Initialize new calculator that only considers k-space in xy-plane,
67 # since we're only looking at the surface
68 calc2 = GPAW(mode=PW(energy), # use the LCAO basis mode
69              h=0.18, # grid spacing
70              xc='PBE', # XC-functional
71              mixer=mixer,
72              kpts=(12, 12, 1), # k-point grid
73              txt='out2.txt') # name of GPAW output text file
74
75 surface111.set_calculator(calc2)
76 surface100.set_calculator(calc2)
```

```

77 cell111 = surface111.get_cell() # Unit cell object of the Al FCC 111
78 area111 = np.linalg.norm(np.cross(cell111[0], cell111[1])) # Calc. surface ←
79     area
80 surfEn111 = surface111.get_potential_energy() # Calc pot. energy of FCC 111
81 cell100 = surface100.get_cell() # Unit cell object of the Al FCC 100
82 area100 = np.linalg.norm(np.cross(cell100[0], cell100[1])) # Calc. surface ←
83     area
84 surfEn100 = surface100.get_potential_energy() # Calc pot. energy of FCC 100
85
86 # Calc. surf. energy per area (sigma) for FCC 111 and 100
87 sigma111 = (1 / (2.0 * area111)) * (surfEn111 - N_x * N_y * E_bulk)
88 sigma100 = (1 / (2.0 * area100)) * (surfEn100 - N_x * N_y * E_bulk)
89
90 ### Add CO adsorbate to Al surface ###
91 d_CO = 1.128 # CO bondlength in [ ]
92
93 CO = Atoms('CO') # Create CO molecule object
94 add_adsorbate(slab=surface111, adsorbate=CO, height=4.5, position='ontop')
95 add_adsorbate(slab=surface100, adsorbate=CO, height=4.5, position='ontop')
96 # height above based on values for CO in ASE doc. Future: We could also
97 # perform equilibrium scan by looping over various heights
98
99 CO.set_cell([10, 10, 10])
100 CO.center()
101 CO.set_calculator(calc)
102
103 energy_CO = CO.get_potential_energy()
104
105 surfEn111_ads = surface111.get_potential_energy()
106 surfEn100_ads = surface100.get_potential_energy()
107
108 sigma100_ads = sigma100 + theta * (surfEn100_ads - surfEn100 - energy_CO) / ←
109     area100
110 sigma111_ads = sigma111 + theta * (surfEn111_ads - surfEn111 - energy_CO) / ←
111     area111
112
113 file = open('sigma_Al.txt', 'w')
114 file.write(str(sigma111) + '\t' + str(sigma100))
115 file.close()
116
117 file = open('sigma_ads.txt', 'w')
118 file.write(str(sigma111_ads) + '\t' + str(sigma100_ads))
119 file.close()
120
121 if rank == 0:
122     a = 0
123     print area111, surfEn111, E_bulk, sigma111

```

A.2 Data processing and wulff construction: HA2_data_processing.py

```

1
2 #from ase.io import read
3 from ase.build import fcc111, fcc100
4 from ase.cluster.wulff import wulff_construction
5 from ase.visualize import view
6 import matplotlib.pyplot as plt
7 import numpy as np
8
9 # input files
10 filename1 = 'sigmas.txt'
11 filename2 = 'sigmas_ads.txt'
12
13 # import and manage data
14 sigmas = np.array(np.loadtxt(filename1))
15 sigma111 = sigmas[0]
16 sigma100 = sigmas[1]
17 sigmas_ads = np.array(np.loadtxt(filename2))
18 sigma111_ads = sigmas_ads[0]
19 sigma100_ads = sigmas_ads[1]
20 theta = 1 / 4 # Monolayer coverage
21 print(sigma111, sigma100)
22 print(sigma111_ads, sigma100_ads)
23
24 Al = wulff_construction('Al',
25     surfaces=[(1, 0, 0),
26               (1, 1, 1)],
27     energies=[sigma100, sigma111],
28     size=10000,
29     structure='fcc',
30     rounding='below') # What does this one do?
31
32 Al.center(vacuum=10)
33 view(Al)
34
35 sigma100_ads = sigma100 + theta * sigma100_ads
36 sigma111_ads = sigma111 + theta * sigma111_ads

```

```
36 Al_ads = wulff_construction('Al',
37                             surfaces=[(1, 0, 0),
38                                       (1, 1, 1)],
39                             energies=[sigma100_ads, sigma111_ads],
40                             size=10000,
41                             structure='fcc',
42                             rounding='below') # What does this one do?
43
44 Al_ads.center(vacuum=10)
45 view(Al_ads)
```