

**DFTB<sup>+</sup>XT**

**open software package for quantum nanoscale modeling\***

**USER MANUAL**

Version 1.02 (Development) July 12, 2018

\*<http://quantranspro.org/dftb+xt/>



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# Preface

**DFTB<sup>+</sup>XT** is a general open source software package for

- fast atomistic electronic structure and molecular dynamics simulations
- model and atomistic quantum transport at nanoscale
- many-body nonequilibrium phenomena
- material and device modeling

**DFTB<sup>+</sup>XT** is an extended version of the **DFTB<sup>+</sup>** code.

DFTB<sup>+</sup>XT package [1] is based on the DFTB<sup>+</sup> [2, 3] source code. Additionally, it suggests a number of new features, which are in the testing phase and may be included into the DFTB<sup>+</sup> release later. The extended functionality of DFTB<sup>+</sup>XT is mainly focused on many-body quantum transport and applications in nanoscience, material and device modeling.

In this version we present the following new features.

- Model Hamiltonians for transport calculations

We introduced the possibility to read model Hamiltonians from external data files and use it with or without a geometry structure. This is especially important for many-body quantum transport problems. See Sec. 3.6.

- Elastic dephasing

Two models of elastic dephasing ("Büttiker probe" and "vibronic dephasing") can be used now to include the dephasing and dissipation beyond the coherent Green function method. Thus, we made a new step towards realistic material and device modeling. See Sec. 3.7.

- Application to STM spectroscopy

We added new options to simplify and make faster the calculation of currents for systems with changeable geometry (like the STM setup). We also supply the python scripts for modeling of the scanning process over tip position and voltage. See Sec. 3.8.





# Chapter 1

## Introduction

The code DFTB<sup>+</sup> is the Fortran 2003 successor of the old DFTB code, which implements the density functional based tight binding approach [4]. The code has been completely rewritten from scratch and extended with various features.

The main features of DFTB<sup>+</sup> are:

- Non-scc and scc calculations (with an expanded range of SCC accelerators)
  - Cluster/molecular systems
  - Periodic systems (arbitrary  $k$ -point sampling, band structure calculations, etc.)
- l-shell resolved calculations possible
- Spin polarised calculations (both collinear and non-collinear spin)
- Geometry and lattice optimisation
- Geometry optimisation with constraints (in xyz-coordinates)
- Molecular dynamics (NVE, NPH, NVT and NPT ensembles)
- Numerical vibrational mode calculations
- Finite temperature calculations
- Dispersion corrections (van der Waals interactions)
- Ability to treat  $f$ -electrons
- Linear response excited state calculations for cluster/molecular systems
- Geometry optimisation and molecular dynamics in singlet and triplet excited states of spin-free molecules
- LDA+U/pSIC extensions
- $L \cdot S$  coupling
- 3rd order on-site corrections (improved H-bonding)
- QM/MM coupling with external point charges (smoothing possible)

- OpenMP parallelisation
- Automatic code validation (autotest system)
- New user friendly, extensible input format (HSD or XML)
- Dynamic memory allocation
- Additional tool for generating cube files for charge distribution, molecular orbitals, etc. (Waveplot)

## Chapter 2

# Input for DFTB<sup>+</sup>

DFTB<sup>+</sup> can read two formats, either XML or the Human-friendly Structured Data format (HSD). If you are not familiar with the HSD format, a detailed description is given in appendix A. The input file for DFTB<sup>+</sup> must be named `dftb_in.hsd` or `dftb_in.xml`. The input file must be present in the working directory. To prevent ambiguity, the parser refuses to read any input if both files are present. After processing the input, DFTB<sup>+</sup> creates a file of the parsed input, either `dftb_pin.hsd` or `dftb_pin.xml`. This contains the user input as well as any default values for unspecified options. The file also contains the version number of the current input parser. You should always keep this file, since if you want to exactly repeat your calculation with a later version of DFTB<sup>+</sup>, it is recommended to use this file instead of the original input. (You must of course rename `dftb_pin.hsd` into `dftb_in.hsd` or `dftb_pin.xml` into `dftb_in.xml`.) This guarantees that you will obtain the same results, even if the defaults for some non specified options have been changed. The code can also produce `dftb_pin.xml` from `dftb_in.hsd` or *vice versa* if required (see section 2.8).

The following sections list properties and options that can be set in DFTB<sup>+</sup> input. The first column of each of the tables of options specifies the name of a property. The second column indicates the type of the expected value for that property. The letters “l”, “i”, “r”, “s”, “p”, “m” stand for logical, integer, real, string, property list and method type, respectively. An optional prefixing number specifies how often (if more than once) this type must occur. An appended “+” indicates arbitrary occurrence greater than zero, while “\*” allows also for zero occurrence. Alternative types are separated by “|”. Parentheses serve only to delimit groups of settings.

Sometimes a property is only interpreted on meeting some condition(s). If this is the case, the third column gives details of the requirement(s). The fourth column contains the default value for the property. If no default value is specified (“-”), the user is required to assign a value to that property. The description of the properties immediately follows the table. If there is also a more detailed description available for a given keyword somewhere else, the appropriate page number appears in the last column.

Some properties are allowed to carry a modifier to alter the provided value (e.g. converting between units). The possible modifiers are listed between brackets ([]) in the detailed description of the property. If the modifier is a conversion factor for a physical unit, only the unit type is indicated (length, energy, force, time, etc.). A list of the allowed physical units can be found in appendix B.

## 2.1 Main input

The input file for DFTB<sup>+</sup> (dftb\_in.hsd/dftb\_in.xml) must contain the following property definitions:

Name	Type	Condition	Default	Page
Geometry	plm		-	<a href="#">12</a>
Hamiltonian	m		-	<a href="#">25</a>

Additionally optional blocks of definitions may be present:

Name	Type	Condition	Default	Page
Driver	m		{}	<a href="#">14</a>
Options	p		{}	<a href="#">51</a>
Analysis	p		{}	<a href="#">52</a>
ExcitedState	p		{}	<a href="#">54</a>
ParserOptions	p		{}	<a href="#">56</a>

**Geometry** Specifies the geometry for the system to be calculated. See p. [12](#).

**Hamiltonian** Configures the Hamiltonian and its options. See p. [25](#).

**Driver** Specifies a geometry driver for your system. See p. [14](#).

**Options** Various global options for the run. See p. [51](#).

**Analysis** Post-run analysis and properties options. See p. [52](#).

**ExcitedState** Calculations in excited state of the system. See p. [54](#).

**ParserOptions** Various options affecting the parser only. See p. [56](#).

## 2.2 Geometry

The geometry can be specified either directly by passing the appropriate list of properties or by using the GenFormat{} method. It is also possible to make model calculations without geometry with NoGeometry{} option.

### 2.2.1 Explicit geometry specification

If the geometry is being specified explicitly, the following properties can be set:

Periodic	l		No
LatticeVectors	9r	Periodic = Yes	-
TypeNames	s+		-
TypesAndCoordinates	(li3r)+		-

**Periodic** Specifies if the system is periodic in all 3 dimensions or is to be treated as a cluster. If set to Yes, property LatticeVectors{} must be also specified.

**LatticeVectors** [*length*] The *x*, *y* and *z* components of the three lattice vectors if the system is periodic.

**TypeNames** List of strings with the names of the elements, which appear in your geometry.

**TypesAndCoordinates** [relative|*length*] For every atom the index of its type in the TypeNames list and its coordinates. If for a periodic system (Periodic = Yes) the modifier relative is specified, the coordinates are interpreted in the coordinate system of the lattice vectors.

Example: Geometry of GaAs:

```
Geometry = {
  TypeNames = { "Ga" "As" }
  TypesAndCoordinates [Angstrom] = {
    1 0.000000 0.000000 0.000000
    2 1.356773 1.356773 1.356773
  }
  Periodic = Yes
  LatticeVectors [Angstrom] = {
    2.713546 2.713546 0.
    0. 2.713546 2.713546
    2.713546 0. 2.713546
  }
}
```

### 2.2.2 GenFormat{}

You can use the generic format to specify the geometry (see appendix C). The geometry specification for GaAs would be the following:

```
Geometry = GenFormat {
  2 S
  Ga As
  1 1 0.000000 0.000000 0.000000
  2 2 1.356773 1.356773 1.356773
  0.000000 0.000000 0.000000
  2.713546 2.713546 0.
  0. 2.713546 2.713546
  2.713546 0. 2.713546
}
```

It is also possible to include the gen-formatted geometry from a file:

```
Geometry = GenFormat {
  <<< "geometry.gen"
}
```

### 2.2.3 NoGeometry{}

The NoGeometry{} option is set by

Geometry = NoGeometry{}

it is used for model calculations without geometry, in this case the Hamiltonian is read from file. See Sec. 3.6.

## 2.3 Driver

The driver is responsible for changing the geometry of the input structure during the calculation. Currently the following methods are available:

{ } Static calculation with the input geometry.

**SteepestDescent{ }** Geometry optimisation by moving atoms along the acting forces. See p. 14.

**ConjugateGradient{ }** Geometry optimisation using the conjugate gradient algorithm. See p. 16.

**gDIIS{ }** Geometry optimisation using the modified gDIIS method. See p. 17.

**SecondDerivatives{ }** Calculation of the second derivatives of the energy (the Hessian). See p. 17.

**VelocityVerlet{ }** Molecular dynamics with the velocity Verlet algorithm. See p. 18.

**Socket{ }** Hands over control to an external program via a socket interface. See p. 24.

### 2.3.1 SteepestDescent{ }

MovedAtoms	(ils)+		1:-1
MaxForceComponent	r		1e-4
MaxSteps	i		200
StepSize	r		100.0
OutputPrefix	s		"geo_end"
AppendGeometries	l		No
Constraints	(1i3r)*	LatticeOpt = No	{ }
LatticeOpt	l	Periodic = Yes	No
FixAngles	l	Periodic = Yes, LatticeOpt = Yes	No
FixLengths	3l	FixAngles = Yes	No No No
Isotropic	l	Periodic = Yes, LatticeOpt = Yes	No
Pressure	r	Periodic = Yes, LatticeOpt = Yes	0.0
MaxAtomStep	r	MovedAtoms $\neq$ None{ }	0.2
MaxLatticeStep	r	Periodic = Yes, LatticeOpt = Yes	0.2
ConvergentForcesOnly	l	SCC = Yes	Yes

**MovedAtoms** Indices of the atoms which should be moved. The atoms can be specified as a mixture of a list of atoms, ranges of atoms and/or the species of atoms. Index ranges are specified as start:end (without white space as one word!), which inclusively selects all atoms between start and end.

MovedAtoms = 1:6

# equivalent to MovedAtoms = { 1 2 3 4 5 6 }

Negative indices can be used to count backwards from the last atom (-1 = last atom, -2 = penultimate atom, etc.):

```
MovedAtoms = 1:-1 # Move all atoms including the last
```

Species names can be used to select all atoms belonging to a given species:

```
MovedAtoms = Ga # select all Ga atoms
```

Various specifiers can be combined together:

```
# Move atoms 1, 2, 3, all Ga atoms, and the last two atoms.
MovedAtoms = 1:3 Ga -2:-1
```

**MaxForceComponent** [*force*] Optimisation is stopped, if the force component with the maximal absolute value goes below this threshold.

**MaxSteps** Maximum number of steps after which the optimisation should stop (unless already stopped by achieving convergence). Setting this value as -1 runs a huge() number of iterations.

**StepSize** [*time*] Step size ( $\delta t$ ) along the forces. The displacement  $\delta x_i$  along the  $i^{\text{th}}$  coordinate is given for each atom as  $\delta x_i = \frac{f_i}{2m} \delta t^2$ , where  $f_i$  is the appropriate force component and  $m$  is the mass of the atom.

**OutputPrefix** Prefix of the geometry files containing the final structure.

**AppendGeometries** If set to Yes, the geometry file in the XYZ-format will contain all the geometries obtained during the optimisation (instead of containing only the last geometry).

**Constraints** Specifies geometry constraints. For every constraint the serial number of the atom is expected followed by the  $x$ ,  $y$ ,  $z$  components of a constraint vector. The specified atom is not allowed to move along the constraint vector. If two constraints are defined for the same atom, the atom will only be able to move normal to the plane containing the two constraining vectors.

Example:

```
Constraints = {
  # Atom one can only move along the z-axis
  1 1.0 0.0 0.0
  1 0.0 1.0 0.0
}
```

**LatticeOpt** Allow the lattice vectors to change during optimisation. MovedAtoms can be optionally used with lattice optimisation if the atomic coordinates are to be co-optimised with the lattice.<sup>1</sup>

**FixAngles** If optimising the lattice, allow only the lengths of lattice vectors to vary, not the angles between them. For example if your lattice is orthorhombic, this option will maintain that symmetry during optimisation.

---

<sup>1</sup>This is functional but not very efficient at the moment.

**FixLengths** If optimising the lattice with `FixAngles = Yes`, allow only the lengths of the specified lattice vectors to vary.

Example:

```
Driver = ConjugateGradient {
  LatticeOpt = Yes
  FixAngles = Yes # Fix angles between lattice vectors
  FixLengths = {Yes Yes No} # Allow only lat. vector 3 to change length
}
```

**Isotropic** If optimising the lattice, allow only uniform scaling of the unit cell. This option is incompatible with `FixAngles`.

**Pressure** [*pressure*] If optimising the lattice, set the external pressure, leading to a Gibbs free energy of the form  $G = E + PV - TS$  being printed as well (the included entropy term is only the contribution from the electrons, therefore this is not the full free energy).

**MaxAtomStep** Sets the maximum possible line search step size for atomic relaxation.

**MaxLatticeStep** Sets the maximum possible line search step size for lattice optimisation. For `FixAngles` or `Isotropic` calculations this is as a fraction of the lattice vectors or the volume respectively.

**ConvergentForcesOnly** If using an SCC calculation, this option controls whether the geometry optimisation will prematurely stop (= Yes) if the SCC cycle does not converge at any geometric step.

### 2.3.2 ConjugateGradient{}

MovedAtoms	(ils)+		1:-1
MaxForceComponent	r		1e-4
MaxSteps	i		200
OutputPrefix	s		"geo_end"
AppendGeometries	l		No
Constraints	(li3r)*		{}
LatticeOpt	l	Periodic = Yes	No
FixAngles	l	Periodic = Yes, LatticeOpt = Yes	No
Isotropic	l	Periodic = Yes, LatticeOpt = Yes	No
Pressure	r	Periodic = Yes	0.0
MaxAtomStep	r	MovedAtoms $\neq$ None{}	0.2
MaxLatticeStep	r	Periodic = Yes, LatticeOpt = Yes	0.2
ConvergentForcesOnly	l	SCC = Yes	Yes

See previous subsection for the description of the properties.



### 2.3.3 gDIIS{}

Alpha	r		0.1
Generations	i		8
MovedAtoms	(ils)+		1:-1
MaxForceComponent	r		1e-4
MaxSteps	i		200
OutputPrefix	s		"geo_end"
AppendGeometries	l		No
Constraints	(1i3r)*		{}
LatticeOpt	l	Periodic = Yes	No
FixAngles	l	Periodic = Yes, LatticeOpt = Yes	No
Isotropic	l	Periodic = Yes, LatticeOpt = Yes	No
Pressure	r	Periodic = Yes	0.0
MaxLatticeStep	r	Periodic = Yes, LatticeOpt = Yes	0.2
ConvergentForcesOnly	l	SCC = Yes	Yes

Specific properties for this method are:

**Alpha** Initial scaling parameter to prevent the iterative space becoming exhausted (this is dynamically adjusted during the run).

**Generations** Number of generations to consider for the mixing.

See previous subsection for the description of the other properties.<sup>2</sup>

### 2.3.4 SecondDerivatives{}

Calculates the second derivatives of the energy (currently only using a numerical differentiation of the forces). The derivatives matrix is written out for the  $i, j$  and  $k$  directions of atoms  $1 \dots n$  as

$$\frac{\partial^2 E}{\partial x_{i1} \partial x_{i1}} \quad \frac{\partial^2 E}{\partial x_{j1} \partial x_{i1}} \quad \frac{\partial^2 E}{\partial x_{k1} \partial x_{i1}} \quad \frac{\partial^2 E}{\partial x_{i2} \partial x_{i1}} \quad \frac{\partial^2 E}{\partial x_{j2} \partial x_{i1}} \quad \frac{\partial^2 E}{\partial x_{k2} \partial x_{i1}} \quad \dots \quad \frac{\partial^2 E}{\partial x_{kn} \partial x_{kn}}$$

into *hessian.out*

**Note:** for supercell calculations, the derivatives are obtained at the  $\mathbf{q} = 0$  point, irrespective of the k-point sampling used.

**Important:** In order to get accurate results for the second derivatives (and the resulting frequencies) you must set a smaller self-consistent tolerance than the default value in the [Hamiltonian{}](#) section. We suggest `SCCTolerance = 1e-7` or better. A less accurate tolerance can yield nonphysical vibrational frequencies.

Atoms	i+lm	1:-1
Delta	r	1e-4

**Atoms** Index of the atoms for which to calculate the second derivatives. The atoms can be specified via indices, index ranges and species. (See [MovedAtoms](#) in section 2.3.1.)

<sup>2</sup>This approach is distinct from section 2.4.1, but uses a related algorithm based on Ref. [5] and comments from P.R.Bridgdon.

**Delta** Step size for numerical differentiation of forces to get the second derivatives of the energy with respect to atomic coordinates.

### 2.3.5 VelocityVerlet{}

The code propagates atomic motion using velocity Verlet dynamics with optional thermostats or barostats to control the temperature and/or pressure. Information is printed out during the simulation every MDRestartFrequency steps, and logged in the file md.out (see appendix 4.5).

MovedAtoms	(ils)+	1:-1	
Steps	i	-	
TimeStep	r	-	
KeepStationary	l	Yes	
Thermostat	m	-	19
OutputPrefix	s	"geo__end"	
MDRestartFrequency	i	1	
Velocities	(3r)*	-	
Barostat	m	Periodic = Yes	21
ConvergentForcesOnly	l	SCC = Yes	Yes
Xlbomd	p	XlbomdFast not set	22
XlbomdFast	p	Xlbomd not set	22
Masses	p		24

**MovedAtoms** List of atoms to move during the MD. (See more detailed description on page 14.)

**Steps** Number of MD steps to perform. In the case of a thermostat using a TemperatureProfile{} the number of steps is calculated from the profile.

**KeepStationary** Remove translational motion from the system.

**TimeStep** [time] Time interval between two MD steps.

**Thermostat** Thermostating method for the MD simulation. See p. 19.

**OutputPrefix** Prefix of the geometry files containing the final structure.

**MDRestartFrequency** Interval that the current geometry and velocities are written to the XYZ format geometry file and md.out (see section 4.5). In the case of SCC MD runs, the charge restart information is also written at this interval overriding RestartFrequency (see section 2.5).

**Velocities** [velocity] Specified atomic velocities for all the atoms of the given structure (including “velocities” for any stationary atoms, which are silently ignored). This option can be used to restart an MD run, but make sure the geometry is consistent with the specified velocities. The easiest way to do this is to copy both from the same iteration of the XYZ file produced in the previous run (**Note:** the velocities printed in the XYZ files are specified in Å/ps, so this should be set in the input). If restarting an SCC MD run, we **strongly suggest** you use ReadInitialCharges, and in particular read charges for the geometry which you use to restart (iterations at which charges are written to disc are marked in the XYZ file, with the most recent being present in charges.bin).

**Barostat** Berendsen method barostat for the MD simulation. See p. 21.

**ConvergentForcesOnly** If using an SCC calculation, this option controls whether the molecular dynamics will prematurely stop (= Yes) if the SCC cycle does not converge at any geometric step. If the option is set to False, forces will be calculated using the non-converged charges and the molecular dynamics continues. In this case you should consider using ForceEvaluation = 'Dynamics' (or ForceEvaluation = 'DynamicsT0') in the Dftb block as it gives more accurate forces for non-converged charges.

**Xlbomd** If present, extended Lagrangian type molecular dynamics is applied to speed up the simulation. For further options within the Xlbomd block see p. 22.

**Masses** If present, over-ride the atomic masses from the Slater-Koster files. See p. 24

### Thermostat

**None{}** No thermostating during the run, only the initial velocities are set according to either a given temperature or velocities, hence an NVE ensemble should be achieved for a reasonable time step.

InitialTemperature	r	-
--------------------	---	---

**InitialTemperature** [energy] Starting velocities for the MD will be created according the Maxwell-Boltzmann distribution at the specified temperature. This is redundant in the case of specified initial velocities.

**Andersen{}** Andersen thermostat [6] sampling an NVT ensemble.

**Note:** Andersen thermostating has a reputation for suppressing diffusion and also prevents accumulation of data for dynamical properties.

Temperature	rlm	-
ReselectProbability	r	-
ReselectIndividually	1	-
AdaptFillingTemp	1	No

**Temperature** [energy] Target temperature of the thermostat. It can be either a real value, specifying a constant temperature through the entire run or the TemperatureProfile{} method specifying a changing temperature. (See p. 21.)

**ReselectProbability** Probability for re-selecting velocities from the Maxwell-Boltzmann distribution.

**ReselectIndividually** If Yes, each atomic velocity is re-selected individually with the specified probability. Otherwise all velocities are re-selected simultaneously with the specified probability.

**AdaptFillingTemp** If Yes, the temperature of the electron filling is always set to the current temperature of the thermostat. (The appropriate tag for the temperature of the electron filling is ignored.)

**Berendsen{}** Berendsen thermostat [7] samples something like an NVT ensemble (but without the correct canonical fluctuations, be aware of the “flying ice cube” problem before using this thermostat [8]).

Temperature	rlm	-
CouplingStrength	r	Timescale not set
Timescale	r	CouplingStrength not set
AdaptFillingTemp	l	No

**Temperature** [*energy*] Target temperature of the thermostat. It can be either a real value specifying a constant temperature through the entire run or the TemperatureProfile{} method specifying a changing temperature. (See p. 21.)

**CouplingStrength** Dimensionless coupling strength for the thermostat (given as  $\Delta t / \tau_t$  in the original paper where  $\Delta t$  is the MD time step). Alternatively Timescale[*time*] can be set directly as the characteristic length of time to damp the temperature towards the target temperature. The CouplingStrength and Timescale options are mutually exclusive.

**AdaptFillingTemp** If Yes, the temperature of the electron filling is always set to the current temperature of the thermostat. (The appropriate tag for the temperature of the electron filling is ignored.)

**NoseHoover{}** Nosé-Hoover chain thermostat [9] sampling an NVT ensemble, currently with the chain coupled to all of the atoms in the system.

Temperature	rlm	-
CouplingStrength	r	-
ChainLength	i	3
Order	i	3
IntegratorSteps	i	1
Restart	m	
AdaptFillingTemp	l	No

**Temperature** [*energy*] Target temperature of the thermostat. It can be either a real value, specifying a constant temperature through the entire run or the TemperatureProfile{} method specifying a changing temperature. (See p. 21, but note that profiles are not well tested with this thermostat yet.)

**CouplingStrength** [*Frequency*] Frequency of oscillation of the thermostating particles (see section 2.5 of Ref. [9]). This is typically related to the highest vibrational mode frequency of the system.

**ChainLength** Number of particles in the thermostat chain.

**Order** and **IntegratorSteps** See section 4.3 of Ref. [9]).

**Restart** Specifies the internal state of the thermostat chain, using three keywords (all three must be present): x{}, v{} and g{} containing the internal chain positions, velocities and forces respectively (these are provided in md.out). See also section 2.3.5.

**AdaptFillingTemp** If Yes, the temperature of the electron filling is always set to the current temperature of the thermostat. (The appropriate tag for the temperature of the electron filling is ignored.)

**TemperatureProfile{}** Specifies a temperature profile during molecular dynamics. It takes as argument one or more lines containing the type of the annealing (string), its duration (integer), and the target temperature (real), which should be achieved at the end of the given period. For example:

```
Temperature [Kelvin] = TemperatureProfile {  # Temperatures in K
  constant      1   10.0  # Setting T=10 K for the 0th MD-step
  linear        500  600.0 # Linearly rising T in 500 steps up to T=600 K
  constant      2000  600.0 # Constant T through 2000 steps
  exponential    500   10.0 # Exponential decreasing in 500 steps to T=10 K
}
```

The annealing method can be constant, linear or exponential, with the duration of each stage of the anneal specified in steps of the driver containing the thermostat. The starting temperature for each annealing period is the final target temperature of the previous period, with the last step of each stage being at the target temperature. Since the initial stage in the temperature profile has no previous step, the default starting temperature is set to 0, but no actual calculation for that temperature is made. In order to start the calculation from a finite temperature for the first annealing period, a constant profile temperature stage with the duration of only one step should be specified as the first step (see the example). The temperatures of the stages are specified in atomic units, unless the Temperature keyword carries a modifier, as in the example above.

### Barostat

Berendsen barostat [7] samples something like an NPH ensemble for MD (but without the correct fluctuations). Options are provided for either isotropic or cell shape changing pressure control. This can also be used in tandem with a thermostat (p. 19) for the NPT ensemble. If the barostat is used, a partial Gibbs free energy is reported in code output, of the form

$$G = E + PV - TS_{\text{electronic}}.$$

This does not include structural entropy, only any electronic entropy. For barostated constant energy simulations (no thermostat in use), the conserved quantity is the sum of the kinetic and Gibbs partial energies.

Pressure	r	-
Coupling	r	Timescale not set
Timescale	r	Coupling not set
Isotropic	l	Yes

**Pressure** [*pressure*] Sets the external target pressure.

**Coupling** Coupling strength for the barostat (given as  $\beta\Delta t/\tau_p$  in the original paper where  $\Delta t$  is the MD time step and  $\beta$  the compressibility, so the coupling is technically dimensioned as reciprocal pressure, but this is usually ignored). Alternatively Timescale[*time*] can be set directly ( $\beta/\tau_p$ ) as the characteristic length of time to damp the pressure. Typically,  $\beta$  is assumed to be either the experimental value or  $\sim 1$ , and Ref. [7] chooses the time scale to be around 10–100 fs. The Coupling and Timescale options are mutually exclusive.

**Isotropic** Should isotropic scaling of the unit cell be used, or can the cell shape vary? There is a slight inconsistency between the standard forms of these scalings in the literature, which is reproduced here, in brief the isotropic case scales the cell volume by a factor proportional to the differences in the instantaneous and expected pressures (i.e., the cube of the cell vectors), while the anisotropic case changes the cell vectors proportional to the difference instead.

### Extended Lagrangian Born-Oppenheimer dynamics

For several systems Born-Oppenheimer molecular dynamics simulations can be significantly sped up by using the extended Lagrangian formalism described in Ref. [10]. The XLBOMD integrator can be used in two different modes:

- **Conventional XLBOMD scheme (Xlbomd):** The extended Lagrangian is used to predict the input charge distribution for the next time step, instead of taking charges that were converged for the geometry in the previous time step. The predicted starting charges should then require fewer SCC iterations to converge.
- **Fast XLBOMD scheme, XlbomdFast (one diagonalisation per time step):** The extended Lagrangian is used to predict the population for each time step. This predicted population is then used to build the Hamiltonian, but in contrast to the conventional XLBOMD scheme, there is no self consistent cycle – the forces are calculated immediately after the diagonalisation of the first Hamiltonian. The fast XLBOMD method usually only works for systems without SCC instabilities (e.g. wider gap insulators or molecules without degenerate states). See Ref. [10] for details.

The extended Lagrangian dynamics can be activated by specifying either the Xlbomd or the XlbomdFast option block. Both methods offer the following options:

IntegrationSteps	i	5
PreSteps	i	0

**IntegrationSteps** Number of time steps used for determining the population for the next time step. Currently, only integration schemes for 5, 6 or 7 steps are implemented.

**PreSteps** Number of molecular dynamics time steps before the XLBOMD integration becomes activated.

**Note:** At the step where the XLBOMD integrator becomes active, it is initialised with results of several subsequent converged MD steps, so a further IntegrationSteps + 1 steps will be carried out before the extended Lagrangian dynamics starts to predict the charges and accelerate the calculation.

The conventional Xlbomd block has the following specific options in addition to the common XLBOMD settings above:

MinSccliterations	i	1
MaxSccliterations	i	200
ScdTolerance	r	1e-5

**MinSccliterations** Minimum number of SCC iterations to perform at each time step during the extended Lagrangian dynamics.

**MaxSccliterations** Maximum number of SCC iterations to perform at each step in the extended Lagrangian dynamics. If this number of SCC iterations have been reached the forces will be calculated and the MD advances to the next time step. See the note in section 2.4.7 regarding non-convergent k-point sampling.

**SccTolerance** SCC convergence tolerance during the extended Lagrangian dynamics. Once this tolerance has been achieved the SCC cycle will stop and the forces will be calculated. You can use this parameter to override the SccTolerance parameter in the DFTB block for time steps where the extended Lagrangian integrator is active (This way, you can calculate populations with tight SCC tolerance when initialising the XLBOMD integrator, then use a less strict SCC tolerance once the integrator is active).

The XlbomdFast block allows has the following specific options in addition to the common XLBOMD settings above:

TransientSteps	i	10
Scale	r	1.0

**TransientSteps** Enables a smoother transition between Born-Oppenheimer and extended Lagrangian dynamics by carrying out intermediate additional steps with full SCC convergence, during which the converged population and the one predicted by the extended Lagrangian integrator are averaged.

**Scale** Scaling factor for the predicted charge densities  $\in (0, 1]$ . The optimal value is system dependent. One should take the highest possible value that still produces stable dynamics (good conservation of energy).

Example for conventional XLBOMD:

```
Xlbomd {
  IntegrationSteps = 6
  MinSccliterations = 2
  MaxSccliterations = 200
  SccTolerance = 1e-6
}
```

Fast (SCC-free) XLBOMD with one diagonalisation per time step:

```
XlbomdFast {
  PreSteps = 5
  TransientSteps = 10
  IntegrationSteps = 5
  Scale = 0.5
}
```

#### Points to be aware of:

- The extended Lagrangian (especially in the fast mode) needs special force evaluation giving more accurate forces for non-convergent charges. Therefore you must set the ForceEvaluation option to 'Dynamics' (for simulations with finite electronic temperature) or to 'DynamicsT0' (for simulations at 0 K electronic temperature) in the DFTB block (see p. 51).
- The extended Lagrangian implementation only works for the  $(N, V, E)$  ensemble so far, so neither thermostats nor barostats are allowed.
- The extended Lagrangian implementation currently cannot be used for spin-polarised or spin-orbit systems, or when electron filling methods other than Fermi{} filling (see p. 37) are used.

## Masses

Provides values of atomic masses for specified atoms, ranges of atoms or chemical species. This is useful for example to set isotopes for specific atoms in the system.

Mass    p
-----------

Any atoms not given specified masses will use the default values from the appropriate homonuclear Slater-Koster file. An example is given below

```
Masses {
  Mass {
    Atoms = 1:2
    MassPerAtom [amu] = 2.0
  }
}
```

where Atoms specifies the atom or atoms which each have a mass of MassPerAtom assigned.

### 2.3.6 Socket{}

The code tries to connect to a socket interface to receive control instructions from an external driver code.

File	s	Host not set	-
Prefix	s	Host not set	"/tmp/ipi_" for Protocol = i-PI{}
Host	s	File not set	-
Port	i	File is set	-
Verbosity	i		0
Protocol	m		i-PI{}
MaxSteps	i		200

**File** Name of UNIX style file socket to connect to.

**Prefix** Prefix to the file name, in the case of i-PI dialogue, the defaults to the path and file start that i-PI expects.

**Host** Name or ip address of internet host to connect to ("localhost" also possible).

**Port** Port of host to connect to.

**Verbosity** Level of port traffic to document.

**Protocol** Choice of message protocol over the socket connection (only communication with i-PI [11] is currently supported).

**MaxSteps** Number of geometry steps before termination of the DFTB<sup>+</sup> instance. Setting this value as -1 runs a huge() number of iterations.

#### Examples

First an ip address connection:



```
Driver = Socket {
  Host = localhost
  Port = 21012 # port number
  Verbosity = 0 # minimal verbosity
  Protocol = i-PI {} # i-PI interface
  MaxSteps = -1 # Run indefinitely
}
```

Then a UNIX socket via the /tmp file system

```
Driver = Socket {
  File = "dftb" # The protocol defines a default path in this case
  Protocol = i-PI {} # i-PI interface
  MaxSteps = 1000 # Terminate this instance after 1000 steps
}
```

## 2.4 Hamiltonian

For calculations without geometry (if `Geometry = NoGeometry{}`), the type of the Hamiltonian must be set to `Model{}`:

```
Hamiltonian = Model{}
```

The properties of the `Model{}` method are discussed in Sec. 3.6.

Currently only a DFTB Hamiltonian is implemented for *ab initio* atomistic calculations, so you must set `Hamiltonian = DFTB{}` or `Hamiltonian = Model{}`.

The DFTB<sup>+</sup> method may contain the following properties:

SCC	l		No	
SCCTolerance	r	SCC = Yes	1e-5	
MaxSCCIterations	i	SCC = Yes	100	
EwaldParameter	r	Periodic = Yes SCC = Yes	0.0	
OrbitalResolvedSCC	l	SCC = Yes	No	
Mixer	m	SCC = Yes	Broyden{}	30
MaxAngularMomentum	p		-	
Charge	r		0.0	
SpinPolarisation	m	SCC = Yes	{}	32
SpinConstants	p	SpinPolarisation $\neq$ {}	-	34
ShellResolvedSpin	l	OrbitalResolvedSCC = No	No	
SpinOrbit	m	SpinPolarisation $\neq$ Colinear{}	{}	36
Eigensolver	m		RelativelyRobust{}	36
Filling	m		Fermi{}	37
SlaterKosterFiles	plm		-	38
OldSKInterpolation	l		No	
PolynomialRepulsive	plm		{}	
KPointsAndWeights	(4r)+lm	Periodic = Yes	-	39
OrbitalPotential	m	SpinPolarisation $\neq$ {}	{}	41
ReadInitialCharges	l	SCC = Yes	No	
InitialCharges	p	SCC = Yes	{}	
ElectricField	p	SCC = Yes	{}	41
Dispersion	m		{}	43
HBondCorrection	m	SCC = Yes	None {}	48
ThirdOrder	l	SCC = Yes	No	
ThirdOrderFull	l	SCC = Yes	No	47
HubbardDerivs	p	ThirdOrder(Full) = Yes	-	
Differentiation	m		FiniteDiff	50
ForceEvaluation	s		"Legacy"	
CustomisedHubbards	p			
Dephasing	p			72

**SCC** If set to Yes, a self consistent charge (SCC) calculation is made.

**SCCTolerance** Stopping criteria for the SCC. Specifies the tolerance for the maximum difference in any charge between two SCC cycles.

**MaxSCCIterations** Maximal number of SCC cycles to reach convergence. If convergence is not reached after the specified number of steps, the program stops. However in cases where the calculation is not for a static structure (so Driver  $\neq$  {}), this behaviour can be overridden using ConvergentForcesOnly.

**EwaldParameter** Sets the dimensionless parameter  $\alpha$  in the Ewald electrostatic summation for periodic calculations. This controls the fraction of the Ewald summation occurring in real and reciprocal space. Setting it to zero or negative activates an automatic determination of this parameter (default and recommended behaviour). Setting it positive forces the code to use the supplied value. This is useful for very asymmetrical unit cells (typically a slab or nanowire with a huge surrounding vacuum region) since the memory demand of DFTB<sup>+</sup> can increase dramatically in these cases (due to storage of a long range real space neighbour list).

To determine a suitable value of  $\alpha$  for such a cell, you should initially reduce the vacuum region and run a test calculation, looking for the value of the automatically chosen Ewald parameter in the standard output. This is then a suitable choice for the original cell with the large vacuum region.

**OrbitalResolvedSCC** If set to Yes, all distinct Hubbard  $U$  values for the different atomic angular momenta shells are used, when calculating the SCC contributions. Otherwise, the value supplied for the  $s$ -shell is used for all angular momenta. Please note, that the old standard DFTB code was *not* orbitally resolved, so that only the Hubbard  $U$  for the  $s$ -shell was used. Please check the documentation of the SK-files you intend to use as to whether they are compatible with an orbitally resolved SCC calculation (many of the biological files do not use orbitally resolved charges), before you switch this option to Yes. Even if the Hubbard  $U$  values for different shells are the same in the SK-files, this flag would still effect your results, since when it is set to Yes, any charge transfer between atomic shells will change the energy of the system compared to when it is set to No.

**Mixer** Mixer type for mixing the charges in an SCC calculation. See p. 30.

**MaxAngularMomentum** Specifies the highest angular momentum for each atom type. All orbitals up to that angular momentum will be included in the calculation. Several main-block elements require  $d$ -orbitals, check the documentation of the SK-files you are using to determine if this is necessary. Possible values for the angular momenta are s, p, d, f.

Example:

```
MaxAngularMomentum = {
  Ga = "p"      # You can omit the quotes around the
  As = "p"      # orbital name, if you want.
}
```

By using the SelectedShells method it is also possible to combine shells from different Slater-Koster files together to treat atoms containing multiple shells with the same angular momentum. The shells to be picked from a particular atom type should be listed without any separating characters. The list of shells of different atom types should be separated by white spaces.

Example:

```
# Defining sps* basis for Si and C by combining sp and s shells from
# Si and Si2, and C and C2, resp.
MaxAngularMomentum = {
  Si = SelectedShells { "sp" "s" }    # Si atom with sps* basis
  C  = SelectedShells { "sp" "s" }    # C atom with sps* basis
}

# Note, that you have to modify the Slater-Koster file definition accordingly
SlaterKosterFiles = {
  Si-Si = "Si-Si.skf" "Si-Si2.skf" "Si2-Si.skf" "Si2-Si2.skf"
  Si-C  = "Si-C.skf" "Si-C2.skf" "Si2-C.skf" "Si2-C2.skf"
  C-Si  = "C-Si.skf" "C-Si2.skf" "C2-Si.skf" "C2-Si2.skf"
  C-C   = "C-C.skf" "C-C2.skf" "C2-C.skf" "C2-C2.skf"
}
```

If for a given atomic type you pick orbitals from more than one species, you must specify an appropriate combinations of file names for the Slater-Koster tables in `SlaterKosterFiles{}`. For every atom type combination  $n_{SK1} \times n_{SK2}$  Slater-Koster files must be defined, where  $n_{SK1}$  and  $n_{SK2}$  are the number species combined to build up the shells of the two interacting atoms. The file names must be ordered with respect to the interacting species, so that the name for the second interacting species is changed first. Above you see an example, where an extended basis with an  $s^*$ -orbital was generated by introducing the new species "Si2" and "C2", containing the appropriate  $s^*$ -orbital for Si and C, resp., as only orbitals.

In the case of multiple Slater-Koster files for a certain interaction, the repulsive data is read from the first specified file (e.g. Si-Si.skf, Si-C.skf, C-Si.skf and C-C.skf in the example above). The repulsive interactions in the other files are ignored. The mass for a certain species is read from the first SK-file for its homo-nuclear interaction.

Non-minimal basis Slater-Koster data may be directly defined in the SK-files in future.

**Charge** Total charge of the system in units of the electron charge. Negative values mean an excess of electrons. If the keyword `FixedFermiLevel` is present (see section 2.4.5), then value specified here will be ignored.

**SpinPolarisation** Specifies if and how the system is spin polarised. See p. 32.

**SpinConstants** Specifies the atom type specific constants needed for the spin polarised calculations, in units of Hartree. See p. 34.

**SpinOrbit** Specifies if the system includes Russel-Saunders coupling. See p. 36

**Eigensolver** Specifies which eigensolver to use for diagonalising the Hamiltonian. See p. 36.

**Filling** Method for occupying the one electron levels with electrons. See p. 37.

**SlaterKosterFiles** Name of the Slater-Koster files for every atom type pair combination. See 38.

**OldSKInterpolation** If set to Yes (strongly discouraged), the look-up tables for the overlap and non-scc Hamiltonian contribution are interpolated with the same algorithm as in the *old* DFTB code. Please note, that the new method uses a smoother function, is more systematic, and yields better derivatives than the old one. This option is present only for compatibility purposes, and may be removed in the future.

**PolynomialRepulsive** Specifies for each interaction, if the polynomial repulsive function should be used. for every pairwise combination of atoms it should contain a logical value, where Yes stands for the use of a polynomial repulsive function and No for a spline. If a specific pair of species is not specified, the default value No is used.

Example:

```
# Use the polynomial repulsive function for Ga-Ga and As-As interactions
# in GaAs
PolynomialRepulsive = {
  Ga-Ga = Yes
  Ga-As = No
  # As-Ga unspecified, therefore per default set to No
  As-As = Yes
}
```

If you want to apply the same setting for all species pairs, you can specify the appropriate logical value as argument of the SetForAll keyword:

```
# Using polynomial repulsive functions for all interactions in GaAs
PolynomialRepulsive = SetForAll { Yes }
```

**KPointsAndWeights** [relative|absolute] Contains the special  $k$ -points to be used for the Brillouin-zone integration. See p. 39. For automatically generated  $k$ -point grids the modifier should not be set.

**OrbitalPotential** Specifies which (if any) orbitally dependant contributions should be added to the DFTB energy and band structure. See p. 41.

**ReadInitialCharges** If set to Yes the first Hamiltonian is constructed by using the charge information read from the file charges.bin.

**InitialCharges** Specifies initial net charges, either for all atoms or for only selected ones. In order to specify it for all atoms, use the keyword AllAtomCharges and supply the net charge for every atom as a list of real values:

```
InitialCharges = {
  AllAtomCharges = { # Specifies net charge for each atom in an H2O molecule
    -0.88081627 # charge for atom 1 (O)
    0.44040813 # charge for atom 2 (H1)
    0.44040813 # charge for atom 3 (H2)
  }
}
```

Alternatively you can specify charges individually on atoms or species using the AtomCharge keyword. For every AtomCharge declaration you must provide a net charge and the list of atoms, which should be initialised to that net charge. (You can use the same format for the list of atoms, as described at the MovedAtoms keyword in the section for [SteepestDescent](#)):

```
InitialCharges = { # Specifying charge for various species
  AtomCharge = {
    Atoms = H
    ChargePerAtom = 0.44040813
  }
  AtomCharge {
    Atoms = O
    ChargePerAtom = -0.88081627
  }
}
```

Net charge on atoms not appearing in any AtomCharge specification is set to be zero.

**ElectricField** Specifies an external electric field, arising currently from either an applied field or a distribution of electrostatic charges. See p. 41.

**Dispersion** Specifies which kind of dispersion correction to apply. See p. 43.

**Differentiation** Specifies how to calculate finite difference derivatives in the force routines. See p. 50.

**ForceEvaluation** Decides which expressions are used to calculate the ground state electronic forces. See p. 51. **Note:** all methods give the same final forces when the charges are well converged. For non-converged charges however the resulting forces can differ considerably between methods.

**CustomisedHubbards** Enables overriding of the Hubbard U values for given species. If the option `OrbitalResolvedScc` has been set to Yes, you need to specify one Hubbard U value for each atomic shell in the basis of the given atom type, otherwise only one atomic value is required. For all species not specified in this block, the value(s) found in their respective Slater-Koster files will be used.

**Warning:** This option is for experts only! Overriding values stored in the SK-files may result in **inconsistent results**. Make sure you understand the consequences when using this option.

Example:

```
CustomisedHubbards {
  Si = 0.32 0.24
}
```

**Dephasing** Two models of elastic dephasing ("Büttiker probe" and "vibronic dephasing") can be used now to include the dephasing and dissipation beyond the coherent Green function method. Thus, we made a new step towards realistic material and device modeling. See Sec. 3.7.

### 2.4.1 Mixer

DFTB<sup>+</sup> currently offers the charge mixing methods `Broyden{}`, `Anderson{}`, `DIIS{}` (DIIS accelerated simple mixer) and `Simple{}` (simple mixer).

#### **Broyden{}**

Minimises the error function

$$E = \omega_0^2 \left| G^{(m+1)} - G^{(m)} \right| + \sum_{n=1}^m \omega_n^2 \left| \frac{n^{(n+1)} - n^{(n)}}{|F^{(n+1)} - F^{(n)}|} + G^{(m+1)} \frac{F^{(n+1)} - F^{(n)}}{|F^{(n+1)} - F^{(n)}|} \right|^2,$$

where  $G^{(m)}$  is the inverse Jacobian,  $n^{(m)}$  and  $F^{(m)}$  are the charge and charge difference vector in iteration  $m$ . The weights are given by  $\omega_0$  and  $\omega_m$ , respectively. The latter is calculated as

$$\omega_m = \frac{c}{\sqrt{F^{(m)} \cdot F^{(m)}}}, \quad (2.1)$$

$c$  being a constant coefficient [12].

The `Broyden{}` method can be configured using following properties:

MixingParameter	r	0.2
InverseJacobiWeight	r	0.01
MinimalWeight	r	1.0
MaximalWeight	r	1e5
WeightFactor	r	1e-2

**MixingParameter** Mixing parameter.

**InverseJacobiWeight** Weight for the difference of the inverse Jacobians ( $\omega_0$ ).

**MinimalWeight** Minimal allowed value for the weighting factors  $\omega_m$ .

**MaximalWeight** Maximal allowed value for  $\omega_m$ .

**WeightFactor** Weighting factor  $c$  for the calculation of the weighting factors  $\omega_m$  in (2.1).

Note: As the Broyden-mixer stores a copy of the mixed quantity for each SCC iteration at a given geometry, you may consider to choose a different mixer with lower memory requirements, if your system needs density matrix mixing (e.g. DFTB+U), is large and needs a high number of SCC-iterations (MaxSCCIteration).

### Anderson{}

Modified Anderson mixer [13].

MixingParameter	r	0.05
Generations	i	4
InitMixingParameter	r	0.01
DynMixingParameters	(2r)*	{}
DiagonalRescaling	r	0.01

**MixingParameter** Mixing parameter.

**Generations** Number of generations to consider for the mixing. Setting it too high can lead to linearly dependent sets of equation.

**InitMixingParameter** Simple mixing parameter used until the number of iterations is greater or equal to the number of generations.

**DynMixingParameters** Allows specification of different mixing parameters for different levels of convergence during the calculation. These are given as a list of tolerances and the mixing factor to be used below this level of convergence. If the loosest specified tolerance is reached, the appropriate mixing parameter supersedes that specified in MixingParameter.

**DiagonalRescaling** Used to increase the diagonal elements in the system of equations solved by the mixer. This can help to prevent linear dependencies occurring during the mixing process. Setting it to too large a value can prevent convergence. (This factor is defined in a slightly different way from Ref. [13]. See the source code for more details.)

Example:

```
Mixer = Anderson {
  MixingParameter = 0.05
  Generations = 4
  # Now the over-ride the (previously hidden) default old settings
  InitMixingParameter = 0.01
  DynMixingParameters = {
    1.0e-2 0.1 # use 0.1 as mixing if more converged that 1.0e-2
    1.0e-3 0.3 # again, but 1.0e-3
```

```

    1.0e-4 0.5 # and the same
  }
  DiagonalRescaling = 0.01
}
```

### DIIS{}

Direct inversion of the iterative space is a general method to acceleration iterative sequences. The current implementation accelerates the simple mix process.

InitMixingParameter	r	0.2
Generations	i	6
UseFromStart	l	Yes

**MixingParameter** Mixing parameter.

**Generations** Number of generations to consider for the mixing.

**UseFromStart** Specifies if DIIS mixing should be done right from the start, or only after the number of SCC-cycles is greater or equal to the number of generations.

### Simple{}

Constructs a linear combination of the current input and output charges as  $(1 - x)q_{\text{in}} + xq_{\text{out}}$ .

MixingParameter	r	0.05
-----------------	---	------

**MixingParameter** Coefficient used in the linear combination.

## 2.4.2 SpinPolarisation

In an SCC calculation, the code currently supports three different choices for spin polarisation; non-SCC calculations are not spin polarised.

### No spin polarisation ({} )

No spin polarisation contributions to the energy or band-structure.

### Colinear{}

Colinear spin polarisation in the  $z$  direction. The following options can be specified:

UnpairedElectrons	r	0
RelaxTotalSpin	l	No
InitialSpins	p	{}

**UnpairedElectrons** Number of unpaired electrons. This is kept constant during the run, unless the RelaxTotalSpin keywords says otherwise.



**RelaxTotalSpin** If set to Yes, a common Fermi-level is used for both spin channels, so that the total spin polarisation can change during run. In this case, the spin polarisation specified using the UnpairedElectrons keyword is only applied at initialisation. If set to No (default), the initial spin polarisation is kept constant during the entire run.

**InitialSpins** Optional initialisation for spin patterns. If this keyword is present, the default code behaviour is that the initial input charge distribution has a magnetisation of 0. Otherwise if it is not present, the initial input charge distribution has a magnetisation matching the number of UnpairedElectrons keyword.

The initial spin distribution for the input charges can be set by specifying the spin polarisation of atoms. For atoms without an explicit specification, a spin polarisation of zero is assumed. The InitialSpins property block must contain either the AllAtomSpins keyword with a list of spin polarisation values for every atom, or one or more AtomSpin blocks giving the spin for a specific group of atoms using the following keywords:

Atoms	(ils)+	-
SpinPerAtom	r	-

**Atoms** Atoms to specify an initial spin value. The atoms can be specified via indices, index ranges and species. (See MovedAtoms in section 2.3.1.)

**SpinPerAtom** Initial spin polarisation for each atom in this InitialSpins block.

For atoms not appearing in any of the SpinPerAtom block, an initial spin polarisation of 0 is set.

Example (individual spin setting):

```
SpinPolarisation = Colinear {
  UnpairedElectrons = 0.0
  InitialSpins = {
    AtomSpin = {
      Atoms = 1:2
      SpinPerAtom = -1.0
    }
    AtomSpin = {
      Atoms = 3:4
      SpinPerAtom = +1.0
    }
  }
}
```

Example (setting all spins together):

```
SpinPolarisation = Colinear {
  UnpairedElectrons = 0.0
  InitialSpins = {
    AllAtomSpins = { -1.0 -1.0 1.0 1.0 } # Atoms 1,2: -1.0, atoms 3,4: 1.0
  }
}
```

### NonCollinear{}

Non-collinear spin polarisation with arbitrary spin polarisation vector on every atom. The only option allowed is to set the initial spin distribution:

InitialSpins	p	}
--------------	---	---

**InitialSpins** Initialisation of the  $x$ ,  $y$  and  $z$  components of the initial spins for atoms. The default code behaviour is an initial spin polarisation of (0 0 0) for every atom.

The initial spin distribution can be set by specifying the spin polarisation vector on all atoms using the AllAtomSpins keyword or by using one or more AtomSpin blocks with the following options:

Atoms	(ils)+	-
SpinPerAtom	(3r)+	-

**Atoms** Atoms to specify an initial spin vector. The atoms can be specified via indices, index ranges and species. (See MovedAtoms in section 2.3.1.)

**SpinPerAtom** Initial spin polarisation for each atom in this InitialSpins block.

For atoms not appearing in any of the SpinPerAtom block, the vector (0 0 0) is set.

Please note, that in contrast to the collinear case, in the non-collinear case you must specify the spin vector (3 components!) for the atoms.

Example:

```
SpinPolarisation = NonCollinear {
  InitialSpins = {
    # Setting the spin for all atoms in the system
    AllAtomSpins = {
      0.408 -0.408 0.816
      0.408 -0.408 0.816
      -0.408 0.408 -0.816
      -0.408 0.408 -0.816
    }
  }
}
```

Example:

```
SpinPolarisation = NonCollinear {
  InitialSpins = {
    AtomSpin = {
      Atoms = 1:2
      SpinPerAtom = 0.408 -0.408 0.816
    }
    AtomSpin = {
      Atoms = 3:4
      SpinPerAtom = -0.408 0.408 -0.816
    }
  }
}
```

## SpinConstants

For each atomic species in the calculation the spin coupling constants for that atom must be specified.

When `OrbitalResolvedSCC = No`, an extra keyword in this block controls whether the spin constants are resolved by shell or are identical for all shells: `ShellResolvedSpin`, defaulting to the same value as `OrbitalResolvedSCC`.

When shell resolved spin constants are specified, they must be ordered with respect to the pairs of shells they couple, such that the index for the second shell increases faster. For an *spd*-basis, that gives the following ordering:

$$w_{ss}, w_{sp}, w_{sd}, \dots, w_{ps}, w_{pp}, w_{pd}, \dots, w_{ds}, w_{dp}, w_{dd}, \dots$$

Example (GGA parameters for H<sub>2</sub>O):

```
SpinConstants = {
  O = {
    # Wss Wsp Wps Wpp
    -0.035 -0.030 -0.030 -0.028
  }
  H = {
    # Wss
    -0.072
  }
}
```

Several standard values of atomic spin constants are given in appendix D. Constants calculated with the same density functional as the SK-files should be used. This input block may be moved to the SK-data definition files in the future.

When using the `SelectedShells` method for the keyword `MaxAngularMomentum`, the spin constants are listed as an array of values running over `SK1SK2...` in the same order as listed for SlaterKoster-Files.

```
SpinConstants = { # not real values, only an example
  Si = {
    # Wss Wsp Wss*
    -0.035 -0.030 -0.01
    # Wps Wpp Wps*
    -0.030 -0.037 -0.02
    # Ws*s Ws*p Ws*s*
    -0.01 -0.02 -0.01
  }
}
```

For cases where `ShellResolvedSpin = No`, the spin constant for the the highest occupied orbital of each atom should be supplied: Example (GGA parameters for H<sub>2</sub>O):

```
SpinConstants = {
  O = {
    #Wpp
    -0.028
  }
  H = {
```

```
# Wss
-0.072
}
}
```

### 2.4.3 SpinOrbit

If present, specifies that  $L \cdot S$  coupling should be included for a calculation. Currently spin unpolarised and non-collinear spin polarisation are supported, but not collinear spin polarisation. For every atomic species present in the calculation the spin-orbit coupling constants,  $\xi$ , for that atom must be specified for all shells present. The constants must be ordered with respect to the list of shells for the given atom.

In the case that the spin-orbit constant for an  $s$  orbital has been set to be a non-zero value the code prints a warning. For periodic systems, use of this keyword automatically prevents the folding by inversion normally used in `SupercellFolding{}`, but manually specified `KPointsAndWeights` should *not* be reduced by inversion.

Example (GaAs):

```
SpinOrbit = {
  Ga [eV] = {0.0 0.12 0.0} # s p d shells
  As [eV] = {0.0 0.32703} # s p shells
}
```

The additional option in this block, `Dual`, sets whether to use a block population for the local spin matrices consistent with the dual populations of Han *et al.* [14] or the conventional on-site part of the single particle density matrix. The default value of this option is `Yes`, also giving extra information regarding atomic orbital moments in the detailed output.

### 2.4.4 Eigensolver

Currently the following LAPACK 3.0 [15] eigensolver methods are available:

- `QR{}`  
(QR decomposition based solver)
- `DivideAndConquer{}`  
(this requires about twice the memory of the other solvers)
- `RelativelyRobust{}`  
(using the subspace form but calculating all states)

None of them needs any parameters or properties specified.

Example:

```
Eigensolver = DivideAndConquer {}
```

### 2.4.5 Filling

There are currently two types of filling supported (see below). Both have common options:

Temperature	r	AdaptFillingTemp = No	0.0
IndependentKFilling	1	Periodic = Yes	No
FixedFermiLevel	(1 2)r		-

**Temperature** [*energy*] Electron temperature in energy units. This property is ignored for thermostated MD runs, if the AdaptFillingTemp property of the thermostat has been set to Yes (See p. 19).

**IndependentKFilling** Causes the occupation of the eigenstates to be independently determined for each  $k$ -point, thus preventing electron transfer between the  $k$ -points. Please note that the value for the Fermi level printed out by the code is meaningless in that case, since there is no common Fermi level for all  $k$ -points. This option is incompatible with use of the FixedFermiLevel keyword.

**FixedFermiLevel** [*energy*] Can be used to fix the Fermi-level (total chemical potential,  $\mu$ ) of the electrons in the system. For collinear spin polarisation, values for up and down spin channels are required. Otherwise only a single global chemical potential is required. If this option is present, the total charge and the total spin of the system are not conserved. (Settings in the options Charge and UnpairedElectrons will be ignored.)

#### Fermi{}

Fills the single particle levels according to a Fermi distribution. When using a finite temperature, the Mermin free energy (which the code prints) should be used instead of the total energy. This is given by  $E - TS$ , where the electron entropy  $S$  is used.

Example:

```
Filling = Fermi {
  Temperature [K] = 300
}
```

#### MethfesselPaxton{}

Produces a Fermi-like distribution but with much lower electron entropy [16]. This is useful for systems that require high electron temperatures (for example when calculating metallic systems). There is an additional option for this type of filling:

Order	i	2
-------	---	---

**Order** Order of the Methfessel-Paxton scheme, the order must be greater than zero, and the 1st order scheme is equivalent to Gaussian filling.

**Note:** Due to the non-monotonic behaviour of the Methfessel-Paxton filling function, the position of the Fermi-level is not necessary unique for a given number of electrons. Therefore, different fillings, band entropies, and Mermin free energies may result, depending which one has been found by the Fermi-level search algorithm. The differences, however, are usually not physically significant.

### 2.4.6 SlaterKosterFiles

There are two different ways to specify the Slater-Koster files for the atom type pairs, explicit specification and using the `Type2FileNames{}` method.

#### Explicit specification

Every pairwise permutation atomic types, connected by a dash, must occur as a property with the name of the corresponding file as an assigned value.

Example (GaAs):

```
SlaterKosterFiles = {
  Ga-Ga = "./Ga-Ga.skf"
  Ga-As = "./Ga-As.skf"
  As-Ga = "./As-Ga.skf"
  As-As = "./As-As.skf"
}
```

If you treat shells from different species as shells of one atom by using the `SelectedShells{}` keyword in the `MaxAngularMomentum{}` block, you have to specify more than one file name for certain species pairs. (For details see the description about the `MaxAngularMomentum{}` keyword.)

#### Type2FileNames{}

You can use this method to generate the name of the Slater-Koster files automatically using the element names from the input geometry. You have to specify the following properties:

Prefix	s	""
Separator	s	""
Suffix	s	""
LowerCaseTypeName	l	No

**Prefix** Prefix before the first type name, usually the path.

**Separator** Separator between the type names.

**Suffix** Suffix after the name of the second type, usually extension.

**LowerCaseTypeName** If the name of the types should be converted to lower case. Otherwise they are used in the same way, as they were specified in the geometry input.

Example (for producing the same file names as in the previous section):

```
SlaterKosterFiles = Type2FileNames {
  Prefix = "./"
  Separator = "-"
  Suffix = ".skf"
  LowerCaseTypeName = No
}
```

The `Type2FileNames` method can not be used if an extended basis was defined with the `SelectedShells` method.

### 2.4.7 KPointsAndWeights

The  $k$ -points for the Brillouin-zone integration can either be specified explicitly or using the `KLines{}` or the `SupercellFolding{}` methods. If the latter is used the `KPointsAndWeights` keyword is not allowed to have a modifier.

#### Explicit specification

For every  $k$ -point four real numbers must be specified: The coordinates of the given  $k$ -point followed by its weight. By default, the coordinates are specified in fractions of the reciprocal lattice vectors. If the modifier `absolute` is set for the `KPointsAndWeights` keyword, absolute  $k$ -point coordinates in atomic units are instead expected. The sum of the  $k$ -point weights is automatically normalised by the program.

```
KPointsAndWeights = { # 2x2x2 MP-scheme
  0.25 0.25 0.25 1.0
  0.25 0.25 -0.25 1.0
  0.25 -0.25 0.25 1.0
  0.25 -0.25 -0.25 1.0
}
```

#### SupercellFolding{}

This method generates a sampling set containing all the special  $k$ -points in the Brillouin zone related to points that would occur in an enlarged supercell repeating of the current unit cell. If two  $k$ -points in the BZ are related by inversion, only one (with double weight) is used (in the absence of spin-orbit coupling this is permitted by time reversal symmetry). The `SupercellFolding{}` method expects 9 integers and 3 real values as parameters:

$$\begin{array}{ccc} n_{11} & n_{12} & n_{13} \\ n_{21} & n_{22} & n_{23} \\ n_{31} & n_{32} & n_{33} \\ s_1 & s_2 & s_3 \end{array}$$

The integers  $n_{ij}$  specify the coefficients used to build the supercell vectors  $\mathbf{A}_i$  from the original lattice vectors  $\mathbf{a}_j$ :

$$\mathbf{A}_i = \sum_{j=1}^3 n_{ij} \mathbf{a}_j.$$

The real values,  $s_i$ , specify the point in the Brillouin-zone of the super lattice, in which the folding should occur. The coordinates must be given in relative coordinates, in the units of the reciprocal lattice vectors of the super lattice.

The original  $l_1 \times l_2 \times l_3$  Monkhorst-Pack sampling [17] for cubic lattices corresponds to a uniform extension of the lattice:

$$\begin{array}{ccc} l_1 & 0 & 0 \\ 0 & l_2 & 0 \\ 0 & 0 & l_3 \\ s_1 & s_2 & s_3 \end{array}$$

where  $s_i$  is 0.0, if  $l_i$  is odd, and  $s_i$  is 0.5 if  $l_i$  is even. For the  $2 \times 2 \times 3$  scheme, you would write for example

```
# 2x2x3 MP-scheme according original paper
KPointsAndWeights = SupercellFolding {
  2  0  0
  0  2  0
  0  0  3
  0.5 0.5 0.0
}
```

To use  $k$ -points for hexagonal lattices which are consistent with the erratum to the original paper [18], you should set the shift for the unique “ $c$ ” direction,  $s_3$ , in the same way as in the original scheme. The  $s_1$  and  $s_2$  shifts should be set to be 0.0 independent of whether  $l_1$  and  $l_2$  are even or odd. So, for a  $2 \times 3 \times 4$  sampling you would have to set

```
# 2x3x4 MP-scheme according modified MP scheme
KPointsAndWeights = SupercellFolding {
  2  0  0
  0  3  0
  0  0  4
  0.0 0.0 0.5
}
```

It is important to note that DFTB<sup>+</sup> does not take the symmetry of your system explicitly into account. For small high symmetric systems with a low number of  $k$ -points in the sampling this could eventually lead to unphysical results. (Components of tensor properties—e.g. forces—could be finite, even if they must vanish due to symmetry reasons.) For those cases, you should explicitly specify  $k$ -points with the correct symmetry.

### **KLines{}**

This method specifies  $k$ -points lying along arbitrary lines in the Brillouin zone. This is useful when calculating the band structure for a periodic system. (In that case, the charges should be initialised from the saved charges of a previous calculation with a proper  $k$ -sampling. Additionally for SCC calculations the number of SCC cycles should be set to 1, so that only one diagonalisation is done using the initial charges.)

The KLines{ } method accepts for each line an integer specifying the number of points along the line segment, and 3 real values specifying the end point of the line segment. The line segments do not include their starting points but their end points. The starting point for the first line segment can be set by specifying a (zeroth) segment with only one point and with the desired starting point as end point. The unit of the  $k$ -points is determined by any modifier of the KPointsAndWeights property. (Default is relative coordinates.)

Example:

```
KPointsAndWeights [relative] = KLines {
  1  0.5 0.0 0.0  # Setting (and calculating) starting point 0.5 0.0 0.0
  10 0.0 0.0 0.0  # 10 points from 0.5 0.0 0.0 to 0.0 0.0 0.0
  10 0.5 0.5 0.5  # 10 points from 0.0 0.0 0.0 to 0.5 0.5 0.5
  1  0.0 0.0 0.0  # Setting (and calculating) a new starting point
  10 0.5 0.5 0.0  # 10 points from 0.0 0.0 0.0 to 0.5 0.5 0.0
}
```



**Note:** Since this set of k-points probably does not correctly integrate the Brillouin zone, the default value of `MaxSccliterations` is set to be 1 in this case.

### 2.4.8 OrbitalPotential

Currently the FLL (fully localised limit) and pSIC [19] (pseudo self interaction correction ) forms of the LDA+U corrections [20] are implemented. These potentials effect the energy of states on designated shells of particular atoms, usually increasing the localisation of states at these sites. The FLL potential lowers the energy of occupied states localised on the specified atomic shells while raising the energy of unoccupied states. The the pSIC potential corrects the local part of the self-interaction error and so lowers the energy of occupied states (see Ref. [19] for a discussion of the relation between these two potentials, and possible choices for the  $UJ$  constant). These particular corrections are most useful for lanthanide/actinide  $f$  states and some localised  $d$  states of transition metals (Ni3d for example).

The Functional option chooses which correction to apply, followed by a list of the specific corrections, listed as an atomic species and the shells on that atom followed by the  $U - J$  constant for that block of shells.

```
OrbitalPotential = {
  Functional = {FLL}
  Si = {
    Shells = {1 2} # sp block on the atom
    UJ = 0.124
  }
}
```

### 2.4.9 ElectricField

This tag contains the specification for an external electric field. Electric fields can only be specified for SCC calculations. You can apply the electric field of point charges<sup>3</sup> and/or a homogeneous external field (which may change harmonically in time). The ElectricField block can currently contain either one or more PointCharges blocks and potentially an External block.

#### PointCharges

The specification for PointCharges has the following properties:

CoordsAndCharges	(4r)+	-
GaussianBlurWidth	r	Periodic = No
		0.0

**CoordsAndCharges** [*length*] Contains the coordinates and the charge for each point charge (four real values per point charge). A length modifier can be used to alter the units of the coordinates. The charge must be specified in proton charges. (The charge of an electron is -1.)

If you read in a huge number of external charges the parsing time to process this data could be unreasonably long. You can avoid this by including the coordinates and the charges directly from an external file via the `DirectRead{}` method (see the example in the next paragraph).

<sup>3</sup>Only in calculations with fixed lattice constants.

Please note that when using this method the program will only read the specified number of records from the external file, and ignores any additional data (so do not leave comments in the external file for example). The external file should contain only one record (3 coordinates and 1 charge) per line.

**GaussianBlurWidth** [*length*] Specifies the half width  $\sigma$  of the Gaussian charge distribution, which is used to delocalise the point charges. The energy of the coulombic interaction  $E_C$  between the delocalised point charge  $M$  with charge  $Q_M$  and the atom  $A$  with charge  $q_A$  is weighted by the error function as

$$E_C(A, M) = \frac{q_A Q_M}{r_{AM}} \operatorname{erf} \left[ \frac{r_{AM}}{\sigma} \right],$$

where  $r_{AM}$  is the distance between the point charge and the atom.

This delocalisation can only be used for non-periodic systems. A length modifier can be used to specify the unit for  $\sigma$ .

Example:

```
ElectricField = {
  # 1st group of charges, with Gaussian delocalisation
  # We have 100000 charges, therefore we choose the fast reading method.
  PointCharges = {
    GaussianBlurWidth [Angstrom] = 3.0
    CoordsAndCharges [Angstrom] = DirectRead {
      Records = 100000
      File = "charges.dat"
    }
  }
}
# 2nd group of charges, no delocalisation (sigma = 0.0)
PointCharges = {
  CoordsAndCharges [Angstrom] = {
    3.3 -1.2 0.9 9.2
    1.2 -3.4 5.6 -3.3
  }
}
```

## External

Specifies a homogeneous external electric field. In the case of *periodic* calculations, a saw-tooth potential is currently used, hence it is up to the user to guarantee that there is a vacuum region isolating periodic copies of the system along the applied field direction. We suggest that you place the structure in the ‘middle’ of the unit cell if possible, to reduce the chances of atoms approaching cell boundaries along the direction of the applied electric field. The code will halt if atoms interact with periodic images of the unit cell along the direction of the electric field.

The External field keyword has the following options

Strength	r	-
Direction	3r	
Frequency	r	molecular dynamics used 0.0
Phase	i	Geometry step offset 0

**Strength** [*Electric field strength*] Specified strength of the applied field.

**Direction** Vector direction of the applied field (the code normalises this vector). In the case of periodic calculations, currently the system *must not* be continuous in this direction (see above).

**Frequency** [*Frequency*] If using molecular dynamics, the field can be time varying with this frequency.

**Phase** Initial field phase in units of geometry steps, this is needed if restarting an MD run in an external field to give the offset in phase of the field after the specified number of steps from the old calculation. The applied field is of the form

$$\mathbf{E}_0 \sin(\omega \Delta t (\text{step} + \text{phase}))$$

where  $\mathbf{E}_0$  is the field vector specified by Strength and Direction,  $\omega$  is the angular Frequency and  $\text{step}$  is the current MD-step in the simulation, using the MD TimeStep of  $\Delta t$  (see section 2.3.5).

### 2.4.10 Dispersion

The Dispersion block controls whether DFTB interactions should be empirically corrected for van der Waals interactions, since DFTB (and SCC-DFTB) does not include these effects. Currently, three different dispersion correction schemes are implemented (for the detailed description of the methods see the following subsections):

- LennardJones: Dispersion is included via a Lennard-Jones potential between each pair of atoms. The parameters for the potential can either be entered by the user or the program can automatically take the parameters from the Universal Force Field (UFF) [21].
- SlaterKirkwood: The dispersion interaction between atoms is taken from a Slater-Kirkwood polarisable atomic model [22].
- DftD3: Dispersion is calculated as in the dftd3 code [23, 24]. Modification hydrogen bond interaction strengths (see section 2.4.12).

#### LennardJones

The Lennard-Jones dispersion model in DFTB<sup>+</sup> follows the method of Ref. [25], using the following potential:

$$\begin{aligned} U_{ij}(r) &= d_{ij} \left[ -2 \left( \frac{r_{ij}}{r} \right)^6 + \left( \frac{r_{ij}}{r} \right)^{12} \right] & r \geq r_0 \\ U_{ij}(r) &= U_0 + U_1 r^5 + U_2 r^{10} & r < r_0 \end{aligned}$$

where  $r_0$  is the distance at which the potential turns from repulsive to attractive. The parameters  $d_{ij}$  and  $r_{ij}$  are built from atomic parameters  $d_i$ ,  $d_j$  and  $r_i$ ,  $r_j$  via the geometrical mean ( $d_{ij} = \sqrt{d_i d_j}$ ,  $r_{ij} = \sqrt{r_i r_j}$ ). The parameters  $U_0$ ,  $U_1$ ,  $U_2$  ensure a smooth functional form at  $r_0$ .

The parameters  $r_i$  and  $d_i$  can either be taken from the parameters of the UFF [21] (as in Ref. [25]) or can be specified manually for each species.

Example using UFF parameters:

```
Dispersion = LennardJones {
  Parameters = UFFParameters {}
}
```

Example using manually specified parameters:

```
Dispersion = LennardJones {
  Parameters {
    H {
      Distance [AA] = 2.886
      Energy [kcal/mol] = 0.044
    }
    O {
      Distance [AA] = 3.500
      Energy [kcal/mol] = 0.060
    }
  }
}
```

The UFF provides dispersion parameters for nearly every element of the periodic table, therefore it can be used for almost all systems “out of the box”. The parameters are also independent of the atomic coordination number, allowing straight forward geometry relaxation or molecular dynamics (unlike the current implementation of Slater-Kirkwood type dispersion).

### SlaterKirkwood

A Slater-Kirkwood type dispersion model is also implemented in DFTB<sup>+</sup> as described in Ref. [22].<sup>4</sup> This model requires atomic polarisation values, van der Waals radii and effective charges for every atom in your system. These parameters are dependent on the coordination of each atom, hence values for different atoms of the same species may vary depending on local environment. You can supply these parameters for the atoms in either of two ways, both using the PolarRadiusCharge tag.

The first option is to specify the values within the PolarRadiusCharge environment by providing three real values (polarisability, van der Waals radius, effective charge) for each atom separately.

Example:

```
Dispersion = SlaterKirkwood {
  # Using Angstrom^3 for volume, Angstrom for length and default
  # unit for charge (note the two separating commas between the units)
  PolarRadiusCharge [Angstrom^3,Angstrom,] = {
    # Polar    Radius    Chrg
    0.560000   3.800000   3.150000   # Atom 1: O
    0.386000   3.500000   0.800000   # Atom 2: H
    0.386000   3.500000   0.800000   # Atom 3: H
  }
}
```

---

<sup>4</sup>Please note, that Ref. [22] contains two typos: equation (7) should read  $C_6^{\alpha\beta} = \frac{2C_6^\alpha C_6^\beta p_\alpha p_\beta}{p_\alpha^2 C_6^\beta + p_\beta^2 C_6^\alpha}$ , in equation (9) the contribution from the dispersion should be  $E_{\text{dis}} = -\frac{1}{2} \sum_{\alpha\beta} f(R_{\alpha\beta}) C_6^{\alpha\beta} (R_{\alpha\beta})^{-6}$ . This option is also currently incompatible with lattice optimisation and the use of barostats.

Alternatively you can provide values for each atomic species in your system, but must supply different values for different coordination numbers using the HybridDependentPol{} keyword. The code needs specific parameters for each type of atom in environments with 0, 1, 2, 3, 4 or  $\geq 5$  neighbours. DFTB<sup>+</sup> then picks the appropriate values for each atom based on their coordination in the *starting* geometry. Two atoms are considered to be neighbours if their distance is less than the sum of their covalent radii, hence you need to supply the covalent radii for each atomic species using the CovalentRadius keyword. This is then followed by a HybridPolarisations block containing a list of six values for atomic polarisabilities then six van der Waals radii and finally a single hybridisation independent effective charge for that atomic species.

Example:

```
Dispersion = SlaterKirkwood {
  PolarRadiusCharge = HybridDependentPol {
    O = {
      CovalentRadius [Angstrom] = 0.8
      HybridPolarisations [Angstrom^3,Angstrom,] = {
        # Atomic polarisabilities 0-5      van der Waals radii 0-5  chrg
        0.560 0.560 0.560 0.560 0.560 0.560  3.8 3.8 3.8 3.8 3.8 3.8  3.15
      }
    }
    H = {
      CovalentRadius [Angstrom] = 0.4
      HybridPolarisations [Angstrom^3,Angstrom,] = {
        # Atomic polarisabilities 0-5      van der Waals radii 0-5  chrg
        0.386 0.396 0.400 0.410 0.410 0.410  3.5 3.5 3.5 3.5 3.5 3.5  0.8
      }
    }
  }
}
```

**Warning:** For both methods of specifying the Slater-Kirkwood dispersion model the code keeps the dispersion parameters fixed for each atom during the entire calculation. Even if the geometry (and therefore the hybridisation) of atoms changes significantly during the calculation, the parameters are unchanged. Therefore if atoms are able to move during your calculation (geometry relaxation or molecular dynamics) you should *always* check whether the coordination of your atoms has changed during the run.

Furthermore, when using the HybridDependentPol{} method we suggest that you first set the StopAfterParsing keyword in the ParserOptions block to Yes (see p. 56) and inspect the generated polarisabilities, radii and charges for every atom in the dftb\_pin.hsd file. If fine tuning of the generated values turns out to be necessary, you should replace the hybrid dependent specification in the input file with corrected atom specific values based on dftb\_pin.hsd.

In order to find suitable parameters for the Slater-Kirkwood model, you should consult Ref. [22] and further references therein. Appendix E contains values which have already been used by some DFTB-users for a few elements.

### DftD3

The DFT-D3 dispersion correction in DFTB<sup>+</sup> is an implementation of the method used in the code 'dftd3' by Stefan Grimme and coworkers. It is based on the *ansatz* described in Refs. [23] and [24].

**Note:** the DFTB<sup>+</sup> binary must be compiled with the DFT-D3 library enabled to use this feature. This dispersion correction for DFTB adds a contribution to the general Kohn-Sham-like energy

$$E_{\text{DFTB-D3}} = E_{\text{DFTB}} + E_{\text{disp}}$$

with  $E_{\text{DFTB}}$  being the DFTB total energy and  $E_{\text{disp}}$  the dispersion energy. The latter contains two-body and optional three-body contributions:

$$E_{\text{disp}} = E_{\text{disp}}^{(2)} + E_{\text{disp}}^{(3)}$$

The form of the two-body contribution can change depending on the chosen damping factor:

- Becke-Johnson damping function:

$$E_{\text{disp}}^{(2)} = -\frac{1}{2} \sum_{A \neq B} \sum_{n=6,8} s_n \frac{C_n^{AB}}{r_{AB}^n + f(R_0^{AB})}$$

with

$$f(R_0^{AB}) = a_1 R_0^{AB} + a_2.$$

- Zero-damping (dispersion at short distances is damped to zero):

$$E_{\text{disp}}^{(2)} = -\frac{1}{2} \sum_{A \neq B} s_n \frac{C_n^{AB}}{r_{AB}^n} f_{d,n}(r_{AB})$$

with

$$f_{d,n} = \frac{1}{1 + 6(r_{AB}/(s_{r,n} R_0^{AB}))^{-\alpha_n}}$$

In order to adjust the dispersion for various energy functionals, the choice of  $s_6$ ,  $s_8$  and the damping parameters  $a_1$  and  $a_2$  (for Becke-Johnson-damping) or  $s_{r,6}$  and  $\alpha_6$  (for zero damping) are treated as functional-dependent values. All other parameters are fixed based on these parameters.

As the DFTB energy functional is largely determined by the underlying parameterisation (the Slater-Koster-files) and the chosen DFTB model (e.g. non-scc, scc, 3rd order, etc.), there are no universal parameter choices which can be used with all settings. The dftd3-program contains optimised parameters for the DFTB3 functional, described in section 2.4.11. The DFTB3 appropriate values are given for the Becke-Johnson-damping function, and are used as default values in the DFTB<sup>+</sup> parser.

**Warning:** Please note, that these default values have only been tested for the DFTB3 model. If you use any other DFTB-model, make sure that the parameters still give reasonable results, and adjust them if needed.

Example using default parameters:

ThirdOrderFull = Yes

DampXH = Yes

Dispersion = DftD3 {}

Example using adjusted parameters with Becke-Johnson damping:

```
Dispersion = DftD3 {
  Damping = BeckeJohnson {
    a1 = 0.5719
    a2 = 3.6017
  }
  s6 = 1.0
  s8 = 0.5883
}
```

Example using zero-damping:

```
Dispersion = DftD3 {
  Damping = ZeroDamping {
    sr6 = 0.7461
    alpha6 = 14.0
  }
  s6 = 1.0
  s8 = 3.209
}
```

### DftD3 optional settings

Apart from the functional dependent dispersion parameters, you can also adjust the additional parameters as shown below. The default values for these parameters are taken to be the same as in the `dftd3` code.

Cutoff	r	$\sqrt{9000}$
CutoffCN	r	40
Threebody	l	No
HHRepulsion	l	No

**Cutoff** [*length*] Cutoff distance when calculating two-body interactions.

**CutoffCN** [*length*] Cutoff distance when calculating three-body interactions.

**Threebody** Whether three-body contributions should be included in the dispersion interactions.

**HHRepulsion** Required when calculating the DFTB3-D3H5 [26] modification to D3 dispersion (see section 2.4.12 for details and parameter values). This keyword enables an additional short range repulsion term in all hydrogen–hydrogen pairs [27] which prevents them from approaching too closely together.

#### 2.4.11 DFTB3

If you would like to use what is called “DFTB3” in some publication(s) [28], this group of options include the relevant modifications to the SCC Hamiltonian and energy. *To enable the DFTB3 model* you will need to set `ThirdOrderFull = Yes` and damp H–X the interactions (see Section 2.4.12).

**ThirdOrder** If set to `Yes` the *on-site* 3rd order correction [29] is switched on. This corrects the SCC-Hamiltonian with the derivatives of the Hubbard U parameters, which you have to specify for every element in `HubbardDerivs`. This correction only alters the on-site elements and is

only maintained for backward compatibility. *You should use the full version `ThirdOrderFull` instead.*

**ThirdOrderFull** If set to Yes the *full* 3rd order correction [28] is switched on. This corrects the SCC-Hamiltonian with the derivatives of the Hubbard U parameters, which you have to specify for every element in `HubbardDerivs`.

**HubbardDerivs** Derivatives of the Hubbard U for the 3rd order correction (on-site or full). For every element the appropriate parameter (in atomic units) must be specified. If you use shell resolved SCC (with full 3rd order), you must specify a list of derivatives for every element, with one Hubbard U derivative for each shell of the given element.

```
Hamiltonian = DFTB {
:
ThirdOrder = Yes
HubbardDerivs {
O = -0.14
H = -0.07
}
:
}
```

## 2.4.12 Hydrogen bond corrections

There are currently two available methods to correct hydrogen bond interactions in the `HBondCorrection` environment:

### Damping

The Damping method modifies the short range contribution to the SCC interaction between atoms *A* and *B* with the damping factor

$$e^{-\left(\frac{U_{Al}+U_{Bl}}{2}\right)^{\zeta} r_{AB}^2}$$

provided that at least one of the two atoms is hydrogen [28, 29]. ( $U_{Al}$  and  $U_{Bl}$  are the Hubbard U values of the two atoms for the *l*-shell,  $r_{AB}$  is the distance between the atoms.) An atom is considered to be a hydrogen-like atom, if its mass (stored in the appropriate homonuclear SK-file) is less than 3.5 amu. The `Exponent` keyword in this environment sets the parameter  $\zeta$  for the short range damping:

```
HBondCorrection = Damping {
Exponent = 4.05
}
```

Table 2 of reference [28] gives suggested values of the exponent for different DFTB2 and DFTB3 models applied to light atoms bonded to hydrogen.

### DFTB3-D3H5

DFTB3-D3H5 [26] is a variant of DFTB3 with additional corrections for non-covalent interactions (dispersion and hydrogen bonds). It consists of a third-order DFTB calculation using the 3OB



parameter set, but where the gamma-function damping (Damping method above) is replaced by the H5 correction and an additional D3 dispersion correction is included. This method also includes a repulsive term which is added to prevent unphysically close approach of pairs of hydrogen atoms [27].

Setting the HBondCorrection environment to H5{} activates this correction for hydrogen bonds [26]. If no additional parameters are provided in the input, suitable values for H-{O,N,S} systems are used (the correction was developed for the DFTB3/3OB model and parameters).

HBondCorrection = H5 {}

**Note:** It was found that DFTB3 overestimates the strength of H-bonds involving the terminal nitrogen of an azide group, and the published results in Ref. [26] were obtained with the H5 correction switched off for these specific atoms. To reproduce this behavior in a system containing nitrogen in several environments, a new atom type with a different name but the same DFTB parameters can be used for specific N atoms to which the correction should not be applied.

If you want to specify the parameters manually, H5 accepts following options, corresponding to terms in Ref. [26]:

RScaling	r	0.714
WScaling	r	0.25
H5Scaling	m	

**RScaling** Global scaling factor,  $s_r$ , when calculating the position of the correcting gaussian functions:

$$r_0 = s_r (r_{\text{vdW}}(X) + r_{\text{vdW}}(H)).$$

**WScaling** Global scaling factor,  $s_w$ , when calculating the width of the correcting gaussian functions. The full-width at half-maximum of the gaussian,  $w$ , is normalised to be 1 for a unit value of WScaling:

$$w = \frac{s_w (r_{\text{vdW}}(X) + r_{\text{vdW}}(H))}{2\sqrt{2\ln 2}}.$$

**H5Scaling** Atom type specific scaling pre-factor,  $k_{\text{XH}}$ , of the correcting gaussian functions when calculating the SCC-interaction:

$$\gamma_{\text{XH}}^{\text{H5}} = \gamma_{\text{XH}} \left( 1 + k_{\text{XH}} \exp \left( -\frac{(r_{\text{XH}} - r_0)^2}{2w^2} \right) \right).$$

You will have to specify one value for each of the chemical species you would like to correct (see the example below). Explicitly setting a negative value (e.g. -1.0) for a given atom type switches off the correction for hydrogen bonds involving that type of atom. In the special cases of N, O or S, if you do not specify a value (and do not disable the contribution by using -1.0), the default value from the reference paper will be used [26]. For any other omitted atom types, the code defaults to a choice of -1.0 (no correction).

```
Hamiltonian = DFTB {
:
HBondCorrection = H5 {
  RScaling = 0.714
  WScaling = 0.25
```

```

H5Scaling {
  O = 0.06
  N = 0.18
  S = 0.21
}
}
:
}

```

**Note:** The van der Waals radii ( $r_{\text{vdW}}$ ) of atoms are also required. DFTB<sup>+</sup> stores these for most of the periodic table, but for cases that are not available their contribution to this correction are neglected.

For a DFTB3-D3H5 calculation, a specific parametrization of the D3 dispersion has to be used. In addition to setting up appropriate values of the D3 parameters, as discussed in Ref. [26], the hydrogen–hydrogen repulsion of Ref. [27] has to also be activated. The complete input is:

```

Hamiltonian = DFTB {
:
  Dispersion = DftD3 {
    Damping = ZeroDamping {
      sr6 = 1.25
      alpha6 = 29.61
    }
    s6 = 1.0
    s8 = 0.49
    HHRepulsion = Yes
  }
:
}

```

### 2.4.13 Differentiation

Calculations of forces currently require the numerical derivatives of the overlap and non-self-consistent Hamiltonian. This environment controls how these derivatives are evaluated.

**Note:** In earlier DFTB<sup>+</sup> versions (up to version 1.2), differentiation was done using finite difference derivatives with a step size of 0.01 atomic units. If you want to reproduce old results, choose the FiniteDiff method and set the step size explicitly to this value.

#### FiniteDiff{}

Finite difference derivatives with a specified step size

Delta	r	epsilon <sup>1/4</sup>
-------	---	------------------------

**Delta** [length] Step size

**Richardson{}**

Extrapolation of finite difference via Richardson's deferred approach to the limit (in principle the most accurate of the currently available choices).

**2.4.14 ForceEvaluation**

Chooses the method for evaluating the electronic contribution to the forces.

'traditional' Uses the "traditional" DFTB-force expression, given for example, in Ref. [30].

'dynamics' Force expression from Ref. [10]. This choice should be used if forces are being calculated with non-converged charges (e.g. when doing XLBOMD dynamics). **Note:** this force expression is only compatible with the Fermi filling (see keyword Filling, p. 37.)

'dynamicsT0' Simplified dynamic force expression valid for electronic temperature  $T = 0$  K [10]. This choice should be used if forces are calculated with non-converged charges and the electronic temperature is zero (e.g. when doing XLBOMD dynamics at  $T = 0$  K).

**Note:** that XLBOMD calculations (Section 2.3.5) are not able to use the 'traditional' forces.

Example:

ForceEvaluation = 'dynamics'

**2.5 Options**

This block collects some global options for the run.

WriteAutotestTag	1	No
WriteDetailedXML	1	No
WriteResultsTag	1	No
WriteDetailedOut	1	Yes
RestartFrequency	i Driver = {}, SCC = Yes	20
RandomSeed	i	0
MinimiseMemoryUsage	1	No
ShowFoldedCoords	1 Periodic = Yes	No
WriteHS	1	No
WriteRealHS	1	No
Verbosity	1	51

**WriteAutotestTag** Turns the creation of the autotest.tag file on and off. (This file can get quite big and is only needed for the autotesting framework.)

**WriteDetailedXML** Turns the creation of the detailed.xml file on and off. (The detailed.xml file is needed among others by the waveplot utility for visualising molecular orbitals.)

**WriteResultsTag** Turns the creation of the results.tag file on and off. (That file is used by several utilities processing the results of DFTB<sup>+</sup>.)

**WriteDetailedOut** Controls the creation of the file `detailed.out`. Since this contains the detailed information about the last step of your run, you shouldn't turn it off without good reasons.

**RestartFrequency** Specifies the interval at which charge restart information should be written to disc for static SCC calculations. Setting it to 0 prevents the storage of restart information. If running an MD calculation, see also section 2.3.5 regarding `MDRestartFrequency`.

**RandomSeed** Sets the seed for the random number generator. The value 0 causes random initialisation. (This value can be used to reproduce earlier MD calculations by setting the initial seed to the same value.)

**MinimiseMemoryUsage** Tries to minimise memory usage by storing various matrices on disc instead of keeping them in memory. Set it to Yes to reduce the memory requirement for calculations with many k-points or spin polarisation.

**ShowFoldedCoords** Print coordinates folded back into the central cell, so if an atom moves outside the central cell it will reappear on the opposite side. The default behaviour is to use unfolded coordinates in the output. (Please note, that this option only influences how the coordinates are printed and written, it does not change the way, periodic systems are treated internally.)

**WriteHS** Instructs the program to build the square Hamiltonian and overlap matrices and write them to files. The output files are `hamsqrN.dat` and `oversqr.dat`, where N enumerates the spin channels. For a detailed description of the file format see p. 77.

**Note:** If either of the options `WriteHS` or `WriteRealHS` are set to Yes, the program only builds the matrices, writes them to disc and then stops immediately. No diagonalisation, no SCC-cycles or geometry optimisation steps are carried out. You can use the `ReadInitialCharges` option to build the Hamiltonian with a previously converged charge distribution.

**WriteRealHS** Instructs the program to build the real space (sparse) Hamiltonian and overlap matrices and write them to files. The output files are `hamreal.dat` and `overreal.dat`. For a detailed description of the file format see p. 77.

**Note:** If either of the options `WriteHS` or `WriteRealHS` are set to Yes, the program only builds the matrices, writes them to disc and then stops immediately. No diagonalisation, no SCC-cycles or geometry optimisation steps are carried out. You can use the `ReadInitialCharges` option to build the Hamiltonian with a previously converged charge distribution.

**Verbosity** This parameter controls the amount of output messages globally and takes values ranging from 1 to 100.

## 2.6 Analysis

This block collects some options to analyse the results of the calculation and/or calculate properties.

<code>AtomResolvedEnergies</code>	1	No	
<code>MullikenAnalysis</code>	1	Yes	
<code>ProjectStates</code>	m	{ }	
<code>Localise</code>	m	{ }	
<code>WriteEigenvectors</code>	1	No	
<code>EigenvectorsAsTxt</code>	1	No	<code>WriteEigenvectors = Yes</code>
<code>WriteBandOut</code>	1	Yes	
<code>CalculateForces</code>	1	No	

**AtomResolvedEnergies** Specifies whether the contribution of the individual atoms to the total energies should be displayed or not.

**MullikenAnalysis** If Yes, the results of a Mulliken analysis of the system is given.

### ProjectStates

ProjectStates evaluates the Mulliken projection of electronic states onto specific regions of the system being modelled (partial density of states – PDOS). The format of the projected data files is similar to band.out, but the second column is the fraction of the state within that region, instead of its occupation number (for non-collinear and spin-orbit calculations, three additional columns for the magnetisation of the state are also given).

Each region for projection is specified within a Region{ } block, with the following options

Atoms	(ils)+	-
ShellResolved	1	No
OrbitalResolved	1	No
Label	s	"region <i>i</i> "

**ShellResolved** Project onto separate atomic shells of the region. These are taken in order of increasing shell number of the atoms. ShellResolved = Yes is only allowed, if all the selected atoms are of the same type.

**OrbitalResolved** Project onto separate atomic orbitals of the region. These are taken in order of increasing shell number of the atoms. As with ShellResolved, this only allowed, if all the selected atoms are of the same type.

**Atoms** Specification of the atoms over which to make the projection.. Atoms are specified in the same way as MovedAtoms in section 2.3.1.)

**Label** Prefix of the label for the resulting file of data for this region. The default is “region*i*.out” where *i* is the number of the region in the input. In the case that ShellResolved = Yes, the shell index is appended, so that files with names “Label.*j*.out” are written. For OrbitalResolved = Yes, the shell and then *m*-value is appended, so that files with names “Label.*j.m*.out” are written.

Examples:

```
ProjectStates = {
  Region = {          # first region
    Atoms = 23:25 27   # atoms 23, 24, 25 and 27
  }
  Region = {
    Atoms = N          # All nitrogen atoms
    ShellResolved = Yes # s and p shells separated instead of atomic PDOS
    Label = "N"         # files N.1.out and N.2.out for s and p states
  }
}
```

### Localise

Convert the single particle states of the calculation to localised orbitals via a unitary transformation. Localised orbitals span the same states as the occupied orbitals, so are equivalent to

the usual valence band states, but are more localised in space. Currently only PipekMezey localisation is supported (but not for non-collinear or spin-orbit calculations).

Pipek-Mezey [31] localisation transforms the occupied orbitals such that the square of the Mulliken charges for each orbital is maximised. The resulting localised states are output as localOrbs.out and localOrbs.bin following the format given in appendix 4.3 for eigenvec.out and eigenvec.bin.

Tolerance	r	1E-4
MaxIterations	i	100

**Tolerance** Cut off for rotations in the localisation process.

**MaxIterations** Maximum number of total sweeps to perform.

For systems with non-gamma-point  $k$ -points, no further options are available.

```
Analysis = {
  Localise = {
    PipekMezey = Yes      # Default options otherwise
    Tolerance = 1.0E-4
  }
}
```

For molecular and gamma point periodic calculations there are two implementations available, Dense = Yes will use the  $O(n^4)$  scaling conventional algorithm, while Dense = No, uses the default sparse method which *may* have better scaling properties.

Dense	I	No
SparseTolerances	r+ Dense = No	1E-1 1E-2 1E-6 1E-12

**Dense** Selects the conventional method (Yes) using Jacobi sweeps over all orbital pairs or (No) uses the default sparse method.

**SparseTolerances** The sparse method introduces support regions during evaluation to increase performance, and these requires a set of tolerances to determine the regions to be used (these are listed in decreasing order, i.e., with tighter tolerances as the localisation proceeds).

**WriteEigenvectors** Specifies, if eigenvectors should be printed in eigenvec.bin. For a description of the file format see p. 78.

**EigenvectorsAsTxt** If eigenvectors are being written, specifies if a text version of the data should be printed in eigenvec.out. For a description of the file format see p. 78.

**WriteBandOut** Controls the creation of the file band.out which contains the band structure in a more or less human friendly format.

**CalculateForces** If Yes, forces are reported, even if not needed for the actual calculation (e.g. static geometry calculation).

## 2.7 ExcitedState

This block collects some options to calculate in the excited state.

Casida	p	SCC = Yes	{}
--------	---	-----------	----

### 2.7.1 Casida

This tag contains the specifications for a time-dependent DFTB calculation, based on linear response theory [32].

**Note:** the DFTB<sup>+</sup> binary must be compiled with linear response calculations enabled to make use of these features (the ARPACK [33] library or ARPACK-ng [34] is required).

The calculation of vertical excitation energies and the corresponding oscillator strengths as well as excited state geometry optimisation can be performed with these options, details of the resulting output files are given in appendix 4.6. Linear response theory is currently implemented only for the SCC-DFTB level of theory and molecular systems.<sup>5</sup> Excitations can be calculated for fractional occupations and collinear spin-polarisation, but forces (and hence geometry optimisation or MD) are only available for spin-unpolarised systems with no fractional occupations. The specifications for this block have the following properties:

NrOfExcitations	i	-
StateOfInterest	i	0
Symmetry	s	SpinPolarisation = {}
EnergyWindow	r	FORTTRAN HUGE()
OscillatorWindow	r	-1
WriteTransitions	l	No
WriteSPTransitions	l	No
WriteMulliken	l	StateOfInterest $\neq$ 0
WriteCoefficients	l	StateOfInterest $\neq$ 0
WriteEigenvectors	l	StateOfInterest $\neq$ 0
TotalStateCoeffs	l	WriteEigenvectors   WriteCoefficients = YesNo
WriteXplusY	l	No
WriteTransitionDipole	l	No
WriteStatusArnoldi	l	No
TestArnoldi	l	No

**NrOfExcitations** Specifies the number of vertical excitation energies to be computed for every symmetry (singlet or triplet). It is recommended that a value slightly greater than the actual number of the states of interest is specified (the eigenvalue solver may not converge to the right roots otherwise).

**StateOfInterest** Specifies the target excited state or states that should be calculated. These are numbered from the first (lowest) excited state as 1, and so on. If the absorption spectrum at a given geometry is required (i.e., a single-point calculation), this parameter should be set to zero (default) and the Driver section (2.3) should be left empty (forces will not be available). A value less than 0 requests that the state with the largest dipole transition moment be found (again a single-point calculation).

**Symmetry** Specifies the spin symmetry of the excited states being computed: “singlet”, “triplet” or “both”. This tag is only applicable for spin restricted calculation. For calculations in the “triplet” or “both” cases, SpinConstants must be supplied (see p. 34).

**EnergyWindow** [*energy*] Energy range above the last transition at NrOfExcitations to be included in excited state spectrum calculation.

<sup>5</sup>Excitation energies can also be calculated for gamma point periodic systems, but will be incorrect for delocalised excitations or for charge transfer-type excited states.

- OscillatorWindow** [*Dipole moment*] Screening cut-off below which single particle transitions are neglected in excitation spectra calculations. This selects from states above the top of the EnergyWindow (if present). This keyword should not be used if calculating forces or other excited state properties.
- WriteTransitions** If set to Yes, the file TRA.DAT is created. This file contains a description of each requested excited state in terms of its single-particle transitions.
- WriteSPTransitions** If set to Yes, the file SPX.DAT is created, which contains the spectrum at the uncoupled DFTB level (i.e. the single-particle excitations).
- WriteMulliken** If set to Yes, the files XCH.DAT and XREST.DAT are created. The former contains atom-resolved Mulliken (net) charges for the excited state of interest, the latter the excited-state dipole moment of the state.
- WriteCoefficients** If set to Yes, the file COEF.DAT is created. This file contains the complex eigenvectors (molecular orbital coefficient) for the excited state of interest. They are derived from the relaxed excited state density matrix.
- WriteEigenvectors** If set to Yes, the file excitedOrbs.bin is created. This file contains the natural orbitals for the specified excited state.
- TotalStateCoeffs** Option to control data from WriteCoefficients or WriteEigenvectors. If set to No the total charge density of the output orbitals corresponds to the change in charge from the ground to excited state. If set to Yes instead it corresponds to the total charge density in the excited state.
- WriteXplusY** If set to Yes, the file XplusY.DAT is created. This file contains the RPA vector  $(X + Y)_{ia}^{I\Sigma}$  for all excited states (c.f., Eqn. (18) in Ref. [35]).
- WriteTransitionDipole** If set to Yes, the file TDP.DAT is created. This file contains the Mulliken transition dipole for each excited state.
- WriteStatusArnoldi** If set to Yes, the file ARPACK.DAT is created, which allows the user to follow the progress of the Arnoldi diagonalisation.
- TestArnoldi** If set to Yes, the file TEST\_ARPACK.DAT is created, which gives data on the quality of the resulting eigenstates.

## 2.8 ParserOptions

This block contains the options, which are effecting only the behaviour of the HSD/XML parser and are not passed to the main program.

ParserVersion	i	current input version
WriteHSDInput	1	Yes
WriteXMLInput	1	No
IgnoreUnprocessedNodes	1	No
StopAfterParsing	1	No

**ParserVersion** Version number of the input parser, which the input file was written for. If you are using an input file, which was created for an older version of DFTB<sup>+</sup>, you should set it to the parser version number of that code version. (The parser version number is printed at the



beginning of the program run to the standard output.) DFTB<sup>+</sup> internally converts the input to its current format. The processed input (written to `dftb_pin.hsd`) is always in the current format, and the `ParserVersion` property in it is always set to be the current parser version.

**WriteHSDInput** Specifies, if the processed input should be written out in HSD format. (You shouldn't turn it off without really good reasons.)

**WriteXMLInput** Specifies, if the processed input should be written out in XML format.

**IgnoreUnprocessedNodes** By default the code stops if it detects unused or erroneous keywords in the input, which probably indicates error(s) in the input. This *dangerous* flag suspends these checks. Use only for debugging purposes.

**StopAfterParsing** If set to Yes, the parser stops after processing the input and written out the processed input to the disc. It can be used to make sanity checks on the input without starting an actual calculation.



## Chapter 3

# Transport calculations

Non-equilibrium Green's function calculations are now possible with DFTB<sup>+</sup>. Within this formalism it is possible to treat quantum mechanical systems with open boundary conditions and therefore quantum transport. A specific new `Transport{}` block has been added to control the transport problem. The NEGF solver parameters can be controlled within the `Eigensolver` section, using the keyword `GreensFunction{}`. Finally the real-space Poisson solver parameters have been organized within a new section, `Electrostatics`, using the keyword `Poisson{}`. The default value for the electrostatic calculations is the usual  $\gamma$ -functional which contains the Hartree and local XC potential.

### 3.1 Definition of the geometry

The input geometry for transport calculations is a little tricky. In comparison to cluster or supercell calculations the geometry for transport calculation must also contain information about the contacts. The contacting leads are actually semi-infinite structures, supporting travelling waves. No stationary current is possible in a finite structure, as travelling waves can only exist in open systems. The structure is partitioned into a device region and two or more contact regions.

Rules to build a valid input structure:

1. All device atoms must come first.
2. Each contact must comprise two subsequent unit cells called principal layers (PLs). The two PLs together give all information about the contact structure and in the following are referred generally as 'contact'.
3. A PL is a unit cell of the contacting lead that has interactions only with nearest neighbour PLs in tight-binding terms.
4. The ordering of the atoms within the two PLs must be consistent in the sense that the two PLs must be exact periodic replica of each other: If each PL comprise N atoms, atom M in the first PL must have a corresponding identical atom in the second PL at position M+N.
5. The first PL should be always the one closer to the device region.
6. All blocks should be contiguous in the structure and each atom must belong to one and only one region.

7. The geometry can be defined as a cluster or a supercell. In the first case is it understood that the contacts are just one-dimensional wire leads.
8. If a structure is defined as *supercell*, only the lattice vectors transverse to the transport direction are meaningful. The periodicity specified along the transport direction is dummy.
9. For each contact the periodicity along the transport direction is deduced from the separation between the two PLs (as the coordinate difference  $\mathbf{r}(M+N) - \mathbf{r}(M)$ ). We refer to this as *contact direction*.
10. All lattice vectors (including the periodicity vector of the contacts) must be aligned to one of the cartesian axes x, y or z. In practice only rectangular cells are currently allowed in transport calculations.

Note: Currently the code makes only a consistency check on the definition of the two PLs (rule 4), namely it checks whether the two contact PLs are really shifted copies of each other. The code does not check if the device regions are consistently defined (rules 1 and 6), if the PL defined are really PLs (rule 3) and does not check if the first PL defined is really the one closest to the device (rule 5). The code checks rules 8, 9 and 10. The check is performed on the atomic coordinates, such that

$$\mathbf{R}_{i+N}^2 = \mathbf{R}_i^1 + \mathbf{v} \quad \forall i \in PL \quad (3.1)$$

where  $\mathbf{R}_i^2$  are atomic coordinates of atoms in the second PL,  $\mathbf{R}_i^1$  are atomic coordinates of atoms in the first PL and  $\mathbf{v}$  is the contact lattice vector. The equality is verified within an accuracy that can be set by the user (see below).

Please take your time to build up structures and cross-check them. Also consider to look at the examples distributed with the code. The input structure is often the first suspect when there's some problem in the calculation.

## 3.2 Transport

DFTB<sup>+</sup> allows for NEGF calculations on different levels. It is possible to calculate linear response transmission within Landauer formalism, charge density out of equilibrium with self-consistent schemes, local currents etc. The Transport section groups the information needed whenever open boundary conditions are used. It contains the description of the partitioning of the system into *device* and *contact* regions and additional contact information needed to calculate the associated Self Energies. The transport block contains the following properties:

Device	p	-	<a href="#">61</a>
Contact	p	-	<a href="#">61</a>
Task	m	UploadContacts	<a href="#">62</a>

Example:

```

Transport {
  Device {
    AtomRange = 1 8
  }
  Contact {
    Id = "source"
  }
}
```

```

    AtomRange = 9 24
  }
  Contact {
    Id = "drain"
    AtomRange = 25 40
  }
  Task = ContactHamiltonian
}

```

### 3.2.1 Device{}

The Device blocks contains the following properties:

Name	Type	Condition	Default	Page
AtomRange	2i		-	<a href="#">62</a>
FirstLayerAtoms	i+		1	
ContactPLs	i+	Geometry = NoGeometry{}	1 1	

**AtomRange** defines the first and last atom of the device region.

**FirstLayerAtoms** defines the first atom of PLs in the device region. By default there is only one layer (the entire device region). Alternatively the user can manually reorder and partition the structure into layers for efficient GF calculations.

Differently from the contact PLs, the device layers do not need to represent unit cell repetitions. The device geometry must be manually ordered in such a way that all the atoms within each layer are contiguous and adjacent layers are placed next to each other. This ensures that the constructed Hamiltonian and Overlap are block tri-diagonal. Refer to [3] for a description of the iterative algorithm.

**ContactPLs** are the indices of PLs coupled to every electrode (e.g. 1 7 means the first contact is coupled to the PL number 1, the second contact – to the PL number 7). Every contact can be coupled to only one of PLs. This property is required for model calculations.

### 3.2.2 Contact{}

The contact block contains the following properties:

Id	s		
AtomRange	2i		
ShiftAccuracy	r		1e-8
FermiLevel	r		
Potential	r		0.0
WideBand	l		No
LevelSpacing	r	WideBand = Yes	20.0

The sections Device and Contact are used to define the atomic range of each region. The user can also assign a label (Id) to each contact that can be used later for cross referencing. In the section Contact the user can add a keyword that specifies the accuracy for the internal check of the PLs.

**Id** Assign a label to each contact.

**AtomRange** Defines the first and last atom of the device region. **Note** that the contacts should be defined in order of increasing atomic range.

**ShiftAccuracy** [*length*] It can be used to set the absolute accuracy used to check the PL consistency (see above). The default is  $10^{-5}$  atomic units. Please be aware that using a large values may hide errors due to a non consistent definition of the contacts, therefore it should not be modified.

**FermiLevel** [*energy*] Specifies the contact Fermi levels.

**Potential** [*energy*] Specifies the electrostatic potential applied to each contact. The natural units of this quantity are potential energy (e.g., V). They can be loosely identified with eV or a.u. since in both these units the electronic charge is practically defined as  $e = 1$ .

**WideBand** Use the wide band approximation for the contact. If set to Yes, the surface green's function of the contact is not explicitly calculated but rather assumed to be local and constant according to a specified density of states.

**LevelSpacing** [*energy*] Specify the inverse of the density of states per atom to be used in the Wide Band approximation. As an example, the DOS of gold at the Fermi level is  $0.05\text{eV}^{-1}\text{atom}^{-1}$ , which corresponds to an energy spacing of  $20\text{ Ev} \simeq 0.735\text{ Hartree}$  (the default value).

### 3.2.3 Task = ContactHamiltonian{}

The Task option is used to define which type of calculation should be performed. Before an SCC transport calculation it is necessary to compute some equilibrium properties of the contacts, by running a periodic boundary conditions DFTB calculation. This necessary step must be carried separately for each contact and can be done by specifying setting Task=ContactHamiltonian as in the following example

```
Task = ContactHamiltonian {
  ContactId = source
  ContactSeparation [Angstrom] = 50.0
}
```

When Task=ContactHamiltonian the following options can be defined

ContactId	s	
ContactSeparation	r	1e3

**ContactId** Id of the contact to be calculated.

**ContactSeparation** [*length*] Dummy separation in transverse direction (see following explanation).

The contact calculation computes the *bulk* Hamiltonian and self-consistent charges for each contact. This is a usual DFTB<sup>+</sup> calculation for which appropriate parameters must be included in the input file. For *supercell* structures the calculation of the contact is performed using corresponding supercells in which the transverse lattice vectors are those specified in the Geometry tag and the lattice vector along the *contact direction* is deduced from the PL separation (rule 9). If the structure is defined as a *cluster*, the contact calculation is performed for a *supercell* in which the contact is treated

as one-dimensional wire. However, since DFTB<sup>+</sup> does not support pure one- and two-dimensional calculations, dummy lattice vectors are defined for the two remaining directions. The default value for these lattice vectors is 1000 a.u. (527 Å), which should guarantee sufficient wire-wire distance to avoid Coulomb interactions. The user can specify an alternative contact separation using the keyword `ContactSeparation` placed in the `ContactHamiltonian` block. Each contact computation produces one output file called `shiftcont_ContactId.dat` storing energy shifts and Mulliken charges that must be present in the working folder in all subsequent transport calculations.

**Note** that during the contact calculation you will need to perform a k-point integration. Whenever the system is defined as a cluster, DFTB<sup>+</sup> will automatically extract the periodicity vectors from the geometry such that the first reciprocal vector will correspond to the transport direction. Therefore you must specify the k-point sampling for the periodic calculation by sampling along the first reciprocal lattice vector. As an example, if the structure is defined as a cluster (i.e., 1-dimensional wire leads), the source contact calculation will have an input file similar to:

```
...
Task = ContactHamiltonian {
  ContactId = source
}
...
Hamiltonian = DFTB {
...
KpointsAndWeights = SupercellFolding {
  8 0 0 # Regardless of transport direction
  0 1 0
  0 0 1
  0.5 0.0 0.0
}
}
```

On the other hand, if your structure is defined as a supercell (as an example, a molecule with bulk contacts) and the transport direction is along y, your the source contact calculation will have an input file similar to:

```
...
Task = ContactHamiltonian {
  ContactId = source
}
...
Hamiltonian = DFTB {
...
KpointsAndWeights = SupercellFolding {
  4 0 0 # Folding in parallel direction
  0 8 0 # Folding in transport direction
  0 0 4 # Folding in parallel direction
  0.5 0.5 0.5
}
}
```

This could seem confusing, but the underlining reasons is that in the cluster calculation the reciprocal lattice is set up by the code itself, while in the periodic calculation is set up by the user who can chose any arbitrary direction. Refer to the transport cookbook and to the distributed examples for further clarification.

### 3.2.4 Task = UploadContacts{}

After the contact calculations, it is possible to perform actual transport calculations. This is activated simply specifying Task = UploadContacts, without additional options. In order to set a proper transport calculation the user should also define the FermiLevel and Potential to each contact.

```
Transport {
  Device {
    AtomRange = 1 8
  }
  Contact {
    Id = "source"
    AtomRange = 9 24
    FermiLevel [eV] = -8.4123
    Potential = 0.0
  }
  Contact {
    Id = "drain"
    AtomRange = 25 40
    FermiLevel [eV] = -8.4123
    Potential = 1.0
  }
  Task = UploadContacts
}
```

Note: During the transport calculation you will not need to set up the k-point integration when the structure is defined as a cluster, just as in a regular DFTB<sup>+</sup> calculation.

## 3.3 GreensFunction

In order to activate Green's functions calculations the user must define the keyword EigenSolver = GreensFunction in the Hamiltonian section. The Green's function section describes all the parameters needed by the Green's function (GF) solver. The GF solver, either under equilibrium (no bias applied) or under non-equilibrium conditions, builds up the density-matrix of the device region. Strictly speaking the GF does not solve for the eigenstates of the open system, however it logically substitutes the traditional construction of the density matrix from the eigenstates of the system, obtained after the diagonalization step. The density matrix can be used to compute any physical observable by 'tracing' with the appropriate operator. In particular it is possible to calculate the Mulliken charges necessary for the DFTB self-consistent loop. Therefore the usual DFTB<sup>+</sup> self-consistent calculations can be driven using the GF solver. The Green's function section contains important parameters used by the solver. The following table describes these parameters.



Name	Type	Condition	Default	Page
Delta	r		1e-5	
ContourPoints	2i		20 20	
LowestEnergy	r		-2.0	
FermiCutoff	i		10	
EnclosedPoles	i		3	
RealAxisStep	r	RealAxisPoints=undefined	6.65e-4	
RealAxisPoints	r	RealAxisStep=undefined		
SaveSurfaceGFs	l		Yes	
ReadSurfaceGFs	l		No	
FirstLayerAtoms	i+	Transport = undefined	1	
FermiLevel	r	Transport = undefined		
LocalCurrents	l		No	

Note: For efficient GF calculation the device region must be partitioned into layers whose fundamental property is to interact with nearest-neighbour layers only.

**Delta** [energy][energy] A small positive imaginary delta used in the GF definition.

**ContourPoints** The number of points along the complex contour integration of the GF along the contour segments  $\mathcal{C}$  and  $\mathcal{L}$  (see contour integration).

**LowestEnergy** [energy][energy] The initial energy from which the integration starts. It should be low enough to ensure that all the electronic states are correctly included in the integration. The default is -2.0 Hartree (see contour integration).

**FermiCutoff** Integer number setting the Fermi distribution cutoff in units of  $kT$ . It is read only if the Fermi distribution temperature is greater than 0 (see contour integration).

**EnclosedPoles** The number of Poles enclosed in the contour. It is meaningful only in finite temperature calculations (see contour integration).

**RealAxisStep** [energy] The energy step along the real axis integration for non-equilibrium calculations. Note: RealAxisStep and RealAxisPoints can not be both defined at the same time.

**RealAxisPoints** The number of points along the real axis integration needed in non-equilibrium calculations. The default depends on the electronic temperature and bias. Note: RealAxisStep and RealAxisPoints can not be both defined at the same time.

**SaveSurfaceGFs** As the SCC cycle usually needs to repeat the calculation of the Green's function at given energy points and as the surface Green functions do not change during the SCC cycle, this flag allows for saving the surface Green functions to disk and save computational time on every SCC cycle beyond the first.

**ReadSurfaceGFs** Allow to load surface Green's function from file also at the first SCC cycle. Note that this operation only makes sense if the energy integration points are identical to the calculation used to generate the surface Green's function files. The code do not verify whether this condition is fulfilled. In general there is no need to modify this default for ReadSurfaceGS and SaveSurfaceGS.

Note: the Green solver can be used also to calculate the density matrix when there is no open boundary conditions, for example to take advantage of the iterative scheme in quasi-1d systems. In this case, a Transport block is not defined and therefore FirstlayerAtoms{}

should be defined here. Also, a Fermi level of the system must be known and provided to fill up the electronic states.

**FirstLayerAtoms** As described in Device block. Can be specified only if no Transport block exists.

**FermiLevel** [*energy*] The Fermi level used by the Green solver to fill up the electronic states.

**LocalCurrents** if set to Yes, local bond-currents are computed using the non-equilibrium density matrix. This task is currently limited to non-periodic systems. The output is placed in a file `lcurrent_u.dat` (or `lcurrent_d.dat` depending on spin). The files are arranged in a table in order of increasing neighbour distance,

Atom(i)	x	y	z	nNeighbors	j1	$I_{i,j1}$	j2	$I_{i,j2}$	j3	$I_{i,j3}$	...
---------	---	---	---	------------	----	------------	----	------------	----	------------	-----

This file can be processed using the small code flux provided in tools/transport that helps in building plots for jmol.

GreensFunction section example:

```
Eigensolver = GreensFunction {
  FirstLayerAtoms = 1 61 92 145
  Delta [eV] = 1e-4
  ContourPoints = 20 20
  RealPoints = 55
  LowestEnergy [eV] = -60.0
  FermiCutoff = 10
  EnclosedPoles = 3
}
```

Note: in order to solve the self-consistent NEGF transport problem, the GreensFunction Eigensolver must be used. However, the TunelingAndDos{} section can be used to calculate the transmission coefficients according to Landauer formula in a non self-consistent manner. In this case an Eigensolver is not needed as the charge density doesn't need to be calculated. To run such a calculation, Eigensolver = TransportOnly must be set and no electrostatics must be calculated (i.e. the Electrostatics = Poisson should not be declared).

### 3.4 Contour integration

Much of the computational work is in the integration of the energy resolved density matrix, represented via the NEGF matrix. The integration is efficiently performed with a complex contour integration and a real axis integration, as shown in Figure 3.1 and discussed in references [36, 37, 3]. All integrations are performed with Gaussian quadratures and the number of points must be specified manually. The complex contour integration is subdivided into two sections: the first section is an arc of circle,  $\mathcal{C}$ , that can be computed with few integration points (default 20); the second section is a line that intersects the contour and runs parallel to the real axis at a distance that depends on the number of poles of the Fermi function enclosed within the contour. Usually a good choice for the number of poles is between 3 and 5 (the default is 3). The poles are placed at the complex points  $z_m = E_F + i(2m + 1)\pi kT$  and, therefore, are separated from each other by  $2\pi kT$ . At  $T = 300$  K this corresponds to a separation of 156 meV. It should be noted that, as the temperature decreases, the separation between poles reduces. This makes the contour integration harder

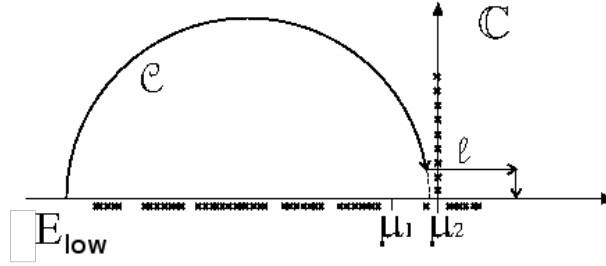


Figure 3.1: Contour integration in the complex plane for the Green's functions. The crosses represent poles of either  $G^r$  or the Fermi function.

as it needs to walk across two singularities. At very low temperatures,  $T = 10$  K, the separation is 5.2 meV. Below this temperature the contour integration is treated as  $T = 0$  in order to avoid numerical inaccuracies. The integration along the segment  $\mathcal{L}$  extends up to  $\text{Re}[z] = E_F + nkT$ , where  $n$  is an integer number specified by the keyword `FermiCutoff` and has a default value of 10. In the limit  $T = 0$  K the poles collapse into a non-analytic cut and the contour needs to be changed such that the second section of the complex contour becomes the arc of circle closing on the real axis. Finally, the real axis integration extends between the lowest and highest chemical potentials. The number of quadrature points should depend on the bias itself and can be set using `RealAxisPoints` or `RealAxisStep`. The default value is 1 pt/0.018 eV (actually 1500 pt/1 H). In finite temperature calculations the segment is extended to include the Fermi cutoff by  $nkT$  on both sides ( $\mu_1 - nkT, \mu_2 + nkT$ ). In this case the number of quadrature points are increased by assuming the same point density defined in the range  $(\mu_1, \mu_2)$ . Example: for a bias of 0.2 V, the default number of points is  $0.2 \cdot 1500 / 27.21139 = 11$ . At  $T = 300$  K the interval is increased by 0.26 eV on both sides, therefore  $0.26 \cdot 1500 / 27.21139 = 14.33$  which is truncated to 14 points, leading to a total of 38 points along the real axis. The use of the keyword `RealAxisStep` is usually more convenient because it ensures a consistent real axis integrations during, for example, a bias sweep.

Note: The GF solver can be used also for calculations other than the transport context. In case the position of the Fermi Energy is known with good accuracy the density matrix solver based on the GF can be used to compute the electronic properties of clusters and supercells. The recursive algorithm may be an efficient solution to large problem and could be efficient for systems having an elongated 1D shape.

### 3.5 Poisson solver

The Poisson solver can not be used at the moment together with `Model NoGeometry` calculations and is not used for `Non-SCC` calculations, but it is a fundamental part of the non-equilibrium SCC transport calculations and must be declared whenever a NEGF calculation is performed using `Electrostatics = Poisson`. Under non-equilibrium conditions the self-consistent potential of the KS equations cannot be solved using the efficient  $\gamma$ -functional, but requires the definition of appropriate boundary conditions for the potentials imposed on the contacts. However, since the  $\gamma$ -functional is formally identical to a pure Hartree potential, it can be obtained in real space by solving a Poisson solver. The Poisson equation is solved in a *box* with hexahedral prism shape. This restriction is imposed by the solver employed. This restricts calculations of supercell structures to orthorhombic super-lattices. An additional restriction is that the box sides must be aligned with the Cartesian axes,  $x, y, z$ .

Name	Type	Condition	Default	Page
PoissonBox	3r			
MinimalGrid	3r		0.5 0.5 0.5	
PoissonAccuracy	r		1e-7	
AtomDensityTolerance	r		1e-6	
CutoffCheck	l		Yes	
Verbosity	i		51	
SavePotential	l		No	
PoissonAccuracy	r		1e-6	
MaxPoissonIterations	i		60	
BuildBulkPotential	l		Yes	69
ReadOldBulkPotential	l		Yes	69
OverrideDefaultBC	m		none{}	69
OverrideBulkBC	m		none{}	69
BoundaryRegion	m		global{}	69
Gate	m		none{}	71
MaxParallelNodes	m		none{}	74

**PoissonBox** [*length*] Dimension of the Poisson box along directions x, y and z.

**MinimalGrid** [*length*] The minimal requested grid spacing along x, y and z. The actual grid spacing chosen by the multigrid will be lower than this. charge densities.

**AtomDensityTolerance** In order to calculate the potential, the Mulliken charges are projected on the real space grid. This parameter defines the cutoff after which the charge is considered vanish (i.e., the space extension of the projected charge). The default is 1e-6. Note that the contacts must be at least twice the length of the extent of a projected Mulliken charge. If this conditions is not fulfilled and is set to Yes, the code will exit with an error message. Setting this parameter to a lower value could allow to define shorter contacts in some cases. However this could lead to relevant error in the potential hence to spurious reflections, therefore it should be left to default value or changed very carefully.

**CutoffCheck** If set to No, consistency between contact length and charge extension is not verified (see above section). The default is No. As for AtomDensityTolerance, this parameter shouldn't be touched unless you know exactly what you're doing.

**Verbosity** This parameter controls the amount of output messages and takes values ranging from 1 to 100.

**SavePotential** Save the potential to file.

**PoissonAccuracy** Defines the accuracy for the approximate solution of the Poisson equation (default value  $10^{-6}$ ).

**MaxPoissonIterations** Defines the maximum amounts of iterations for the solver.

**Note:** The Poisson Box can be specified using the keyword PoissonBox. In calculations in which the two contacts face each other along the same axis, setting the box-size along this axis has no effect because the code needs to adjust the correct size internally. This keyword is redundant (and should not be specified) when the system is periodic, since in this case the Poisson box is taken from the supercell lattice vectors.

Numerical error in the potential will result in spurious discontinuities at the contact-device interfaces. The default tolerances should do the job in most cases.

This is a typical example of the whole Poisson block specification. Some of the keywords are described in the next subsections.

Example:

```
Electrostatics = Poisson {
  PoissonBox [Angstrom] = 20.0 20.0 20.0
  MinimalGrid [Angstrom] = 0.3 0.3 0.3
  SavePotential = No
  BuildBulkPotential = Yes
  ReadOldBulkPotential = No
  BoundaryRegion = Global {}
  PoissonAccuracy = 1e-7
  Gate = Planar{
    GateLength_l [Angstrom] = 10.0
    GateLength_t [Angstrom] = 20.0
    GateDistance [Angstrom] = 7.0
    GatePotential [eV] = 1.0
  }
}
```

### 3.5.1 Boundary Conditions

The Poisson equation is solved imposing special boundary conditions (BC) on the six faces of the Poisson Box. In basic transport calculations, comprising two contacts placed along the same axis, the BCs are chosen as follows:

**Dirichlet** Fixed potentials on the two contact faces with values defined by the applied potentials (see UploadContacts, ContactPotentials).

**Neumann** Zero normal field on the remaining 4 lateral box faces.

In periodic supercells the BCs are: **Dirichlet** (fixed potentials) on the two contact faces with values defined by the applied potentials (see UploadContacts, ContactPotentials) and **Periodic** on the remaining 4 lateral box faces.

In some specific cases Neumann BCs can be set on one contact. In order to do so it is necessary to use OverrideDefaultBC (see below).

Device and contact potentials should smoothly join at the interface. In order to reach this goal the code computes the Bulk potential of each contact and uses the result as a BC on the contact face of the Poisson box. This is useful when the contact potential is not uniform due to charge rearrangements. The external applied contact potential is added to the bulk potential. The user can deactivate this calculation with the keyword BuildBulkPotential.

**Note:** The bulk potential is computed on a special box that has 'lateral' sizes copied from the device box, and has the size of one PL along the contact direction. The BCs are –so to speak– inherited from the device region. In particular:

1. Along the contact direction periodic BCs are imposed.
2. On the other four faces the BCs are copied from the device region.
3. The user can override this setting using `OverrideBulkBC` (see below).
4. When all four faces inherit Neumann BC (default for the device region), these are ALL internally changed to Dirichlet, because the solver cannot handle this situation that gives rise to a singular matrix.

**BuildBulkPotential** (default: Yes) is used to calculate the electrostatic potential of the contacts and the result is used as a Dirichlet boundary condition on the contact face (superimposed to the contact potential).

**ReadOldBulkPotential** Read a previously computed bulk potential from hard-disk.

**BoundaryRegion** Specifies how the Dirichlet boundary conditions are treated on each contact face of the Poisson box. It can be Global, Square or Circle. Global means that the BC is applied to the entire face of the box, whereas the other keywords imply that the Dirichlet BC are applied on a cross-section projected on the contact face. This is useful for instance when handling nanowire contacts, for which it is not really consistent to impose a constant potential on the whole face of the Poisson box.

**BufferLength** [*length*] can be used to set the size of the boundary region beyond the atomistic size which is determined as the minimal circle or square containing all atoms of the contact cross-section.

Example:

```
BoundaryRegion = Circle {
  BufferLength [Angstrom] = 3.0
}
```

In some special case it might be necessary to override the default BCs applied by the code on the Poisson equation. Currently this can be done using the keywords: `OverrideDefaultBC` and `OverrideBulkBC`.

**OverrideDefaultBC** block is used to override the BCs described above. It can be used to force Dirichlet or Neumann BCs along some specified directions or on one of the four lateral faces of the Poisson box.

**Boundaries** is used to specify on which face different BCs must be imposed. Assuming contacts along z, the keyword can be set any of xmin, xmax, x, ymin, ymax, y.

```
OverrideDefaultBC = Dirichlet {
  Boundaries = xmin
}
```

For instance setting Dirichlet BC on `Boundaries = xmin` imposes  $\phi(x, y, z) = 0$  on the face placed at  $x = x_{\min}$ , `boundaries = xmax` imposes  $\phi(x, y, z) = 0$  on the face placed at  $x = x_{\max}$ . When Dirichlet needs to be forced on both faces, it is possible to use either `boundaries = xmin, xmax` or

simply boundaries = x. The same syntax can be used to impose conditions on more faces, using boundaries = x,y or boundaries = x,ymin.

A similar strategy can be used to impose different boundary conditions on the contacts. For instance, a Neumann BC can be set on one contact face using

```
OverrideDefaultBC = Neumann {
    Boundaries = zmin
}
```

**Note** that the user should know which face of the Poisson Box corresponds to the desired contact. Furthermore, if the user sets Neumann at all contacts the Poisson solver will not converge (singular matrix) unless Dirichlet is imposed somewhere else (e.g., a gate potential is present).

It is also possible to override default BCs when computing the bulk potential.

**OverrideBulkBC** block is used to override bulk BC usually copied from the device region.

**Boundaries** has the same meaning and syntax as in OverrideDefaultBC.

```
OverrideBulkBC = Neumann {
    Boundaries = x, y
}
```

### 3.5.2 Electrostatic Gates

The option Gate can be used to specify an electrostatic gate. Currently the gate type Planar and Cylindrical are allowed. Restrictions. The planar gate must be placed with its face parallel to the x-z plane, i.e., the gate direction must be along y. At the same time the transport direction should be placed along the z-axis. The latter is not really a restriction but it gives meaning to 'longitudinal' and 'transverse' in the geometrical definitions of the gate lengths. Example:

```
Gate = Planar {
    GateLength_l [Angstrom] = 20.0
    GateLength_t [Angstrom] = 20.0
    GateDistance [Angstrom] = 7.0
    GatePotential [eV] = 1.0
}
```

```
Gate = Cylindrical {
    GateLength [Angstrom] = 10.0
    GateRadius [Angstrom] = 7.0
    GatePotential [eV] = 1.0
}
```

The various options for the gates have the following meanings:

**GateLength\_l** [*length*] Sets the gate length along the transport direction (assumed to be z). The gate is centered in the device region.

**GateLength\_t** [*length*] Sets the gate transverse to the transport direction (assumed to be x). The gate is centered in the device region.

**GateDistance** [*length*] Sets the distance of the gate from the center axis of the device region.

**GatePotential** [*energy*] Sets the potential applied to the gate.

**GateRadius** [*length*] For cylindrical gate, sets the distance of the gate from the center axis or gate radius.

Note that the gate option has not been tested thoroughly and may still contain bugs. Please report to the developers any problem encountered.

Developments. In forthcoming releases also double gates will be possible. Similarly to a usual DFTB<sup>+</sup> calculation, the output from a Transport calculation will be generated in the `detailed.out` and `detailed.xml`. These files are self-documenting, i.e. you will find a human-readable description of the output data in the files themselves. After a transport calculation, the files will contain the transmission coefficient for every energy point and for every k point and the local density of states for every energy point and projection ranges specified in input. They will also contain the total current and the partial current for every k point. In multiterminal calculation, this data will be written for every terminal couple.

### 3.6 Model Hamiltonians

To use the external Hamiltonian without geometry (if `Geometry = NoGeometry{}`), the type of the Hamiltonian must be set to `Model{}`:

Hamiltonian = `Model{}`

The `Model{}` method may contain the following properties:

Name	Type	Condition	Default	Page
NumStates	i		No default!	
HamiltonianFile	s		H.mtr	

**NumStates** Is a full number of states in the Hamiltonian. Required for model calculations without geometry.

**HamiltonianFile** [*energy*] The name of the data file with the Hamiltonian, saved as an array of real numbers.

Model Hamiltonians are used at the moment only for transport calculations. It is important to add the property `ContactPLs{}` to the `Device{}` section (see Sec. 3.2.1).

### 3.7 Elastic dephasing

To switch on the dephasing effects, the group `Dephasing{}` must be included into the Hamiltonian group. The `Dephasing{}` group may contain the following methods/properties:



Name	Type	Condition	Default	Page
BuettikerProbes	m		none{}	
VibronicElastic	m		none{}	

### 3.7.1 Büttiker probes

**BuettikerProbes** switches on the Büttiker probe dephasing. There are two methods: `ZeroPotential{}` and `ZeroCurrent{}`. The first one describes physical conducting environment with fixed (at the moment zero) electrical potential. The second is for the *Büttiker Probe Model* of dephasing, when the zero-current condition is fulfilled by artificial adjustment of local electrical potentials.

```
BuettikerProbes = ZeroPotential {
  Coupling [eV] = constant { 0.01 }
}
```

```
BuettikerProbes = ZeroCurrent {
  Coupling [eV] = constant { 0.01 }
}
```

**Coupling** [*energy*] describes coupling to the environment. The method `constant{}` couples all states equivalently. The other possible methods are `AtomCoupling{}` – with different couplings for different atoms, and `AllOrbitals{}` – with different couplings for all atomic orbitals. For the model calculations these two options are equivalent.

```
Coupling [eV] = AtomCoupling {
  AtomList { Atoms = 1
    Value = 0.1
  }
  AtomList { Atoms = 2 3
    Value = 0.01
  }
  AtomList { Atoms = 4 5
    Value = 0.03
  }
}
```

```
Coupling [eV] = AllOrbitals { 0.1 0.1 0.1 ... }
```

### 3.7.2 Vibronic dephasing

**VibronicElastic** switches on the elastic vibronic dephasing method. There is only one method `local{}` available at the moment. The `Coupling{}` method can get the same values as for Büttiker probe dephasing. There are several additional options: `AtomBlock` (default=.false.), which changes the way of coupling between electrons and phonons; `MaxNumIter` – the maximum number of iterations (default=100); `Mixing` – control of mixing in the self-consistent cycle (default = 0.05); `Tolerance` – control of the convergency tolerance (default = 0.001).

```

VibronicElastic = local {
  AtomBlock = Yes
  Coupling [eV] = Constant {0.01}
  MaxNumIter = 1000
  Mixing = 0.5
  Tolerance = 0.001
}

```

### 3.8 Application to STM spectroscopy

### 3.9 Parallelizations

The code has been parallelised in two main parts. The Non-equilibrium Green's functions are computed by distributing the energy points along the contour and real axis calculations. Contour and real axis integrations are independent and separately distributed. Load balancing has to be taken care by the user. For instance if ContourPoints = {20 20} and RealAxisPoints = 60, by setting 10 MPI nodes, each node will handle 4 points along the contour and 6 points along the real axis.

Mixed OpenMP/MPI calculations are possible. When compiling dftb+ the user should link against threaded mkl, rather than sequential. Numerical experiments show that best performance on multicore CPUs is generally obtained by running independent MPI processes on physical sockets, exploiting OpenMP multithreading on each socket. For instance NEGF can exploit threaded matrix-matrix products. The user can experiment by setting the environment variable OMP\_NUM\_THREADS.

The Poisson solver has not been parallelized yet. However the efficient multigrid solver does not typically represent a bottleneck. Currently the assembly of the charge density on the real-space grid and the projection of the potential on the atoms have been parallelized. Since gather of the charge density on each node can easily hit communication bottlenecks the user can use the parameter MaxParallelNodes to control distributions of these calculations. The default is MaxParallelNodes=1, that can be increased until speedups are observed.

MaxParallelNodes	i	1
------------------	---	---

### 3.10 Analysis

The Analysis block is used to specify post-scf calculations such as tunneling or projected DOS.

```

Analysis{
  TunnelingAndDOS{
    EnergyRange [eV] = {-5.0 -3.0}
    EnergyStep [eV] = 0.02
  }
}

```

## 3.11 TunnelingAndDos

This method block can be specified in `Analysis{}` and it is used to calculate the transmission by means of Caroli formula, the current by means of Landauer formula and the Density of States from the spectral function. This block can only be specified if an open boundary system has been defined in `Transport{}`.

EnergyRange	2r		
EnergyStep	r		
TerminalCurrents	p		
Region	p		75
WriteTunn	1	Yes	
WriteLDOS	1	Yes	

**EnergyRange** [*energy*] Contains the energy range over which the transmission function and local density of states are computed.

**EnergyStep** [*energy*] Is the energy sampling step.

**TerminalCurrents{}** in multiterminal configurations is used to define the terminal across which current must be computed. The terminal pairs are defined by using the keyword `EmitterCollector`. example:

```
TerminalCurrents{
  EmitterCollector = {"source" "drain"}
  EmitterCollector = {"source" "gate"}
}
```

The block `TerminalCurrents` may be omitted since the code automatically sets all possible independent combinations for the terminal currents. For example in a 4-contact calculations the currents are 1-2, 1-3, 1-4, 2-3, 2-4, 3-4.

**Region{}** This block defines atomic ranges or orbitals where the local density of states is calculated projected. The definition in the block follow the same syntax as a DFTB<sup>+</sup> calculation without transport.

**WriteTunn** The transmission coefficients are written also to a separate file for quick reference. If set to No, the transmission coefficient are only written to DFTB<sup>+</sup> output files (detailed.out and detailed.xml, autotest.tag).

**WriteLDOS** same as above, for the density of states.

## 3.12 Troubleshooting

DFTB<sup>+</sup> transport machinery is designed to calculate transport in structures with a large number of atoms. To take full advantage of the iterative algorithm, be sure that the system is correctly partitioned in Principal Layers, as described in Transport and Green's solver sections. Be aware that a wrong partitioning will lead to wrong results. If you're not completely confident, you can run a calculation on a test system with and without partitioning. The results should be the same.

On some systems, a Segmentation Fault error could occur while running relatively large structures. This could happen because the stack memory limit has been exceeded. You can troubleshoot this setting a higher limit for the stack memory. In bash you can remove stack memory limitation with the command line `ulimit -s unlimited`.

## Chapter 4

# Output of DFTB<sup>+</sup>

This chapter contains the description of some of the output files of DFTB<sup>+</sup> where the output format is not self documenting. Unless indicated otherwise, numbers in the output files are given in atomic units (with Hartree as the energy unit).

### 4.1 hamsqrN.dat, oversqr.dat

The files hamsqrN.dat and oversqr.dat contain the square (folded) Hamiltonian and overlap matrices. The number N in the filename hamrealN.dat indicates the spin channel. For spin unpolarised calculation it is 1, for spin polarised calculation it is 1 and 2 for spin-up and spin-down, respectively while for non-collinear spin it is charge, x, y and z for 1, 2, 3 and 4. Spin orbit is not currently supported for this option.

Only non-comment lines (lines not starting with "#") are documented:

- Flag for signalling if matrix is real (REAL), number of orbitals in the system (NALLORB), number of kpoints (NKPOINT). For non-periodic (cluster) calculations, the number of kpoints is set to 1.
- For every  $k$ -point:
  - Number of the  $k$ -point. For molecular (non-periodic) calculations only 1  $k$ -point is printed.
  - The folded matrix for the given  $k$ -point. It consists of NALLORB lines  $\times$  NALLORB columns. If the matrix is not complex (REAL is F), every column contains two numbers (real and imaginary part).

The files are produced if requested by WriteHS = Yes (see section 2.5).

### 4.2 hamrealN.dat, overreal.dat

The files hamrealN.dat and overreal.dat contain the real space Hamiltonian and overlap matrices. The number N in the filename hamrealN.dat indicates the spin channel. For spin unpolarised calculation it is 1, for spin polarised calculation it is 1 and 2 for spin-up and spin-down, respectively,

while for non-collinear spin it is charge,  $x$ ,  $y$  and  $z$  for 1, 2, 3 and 4. Spin orbit is not currently supported for this option.

Note: The sparse format contains only the "lower triangle" of the real space matrix. For more details about the format and how to obtain the upper triangle elements, see reference [2]. Also note, that for periodic systems the sparse format is based on the *folded* coordinates of the atoms, resulting in translation vectors (ICELL) which look surprising at first glance.

Only non-comment lines (lines not starting with "#") are documented:

- Number of atoms in the system (NATOM)
- For every atom:
  - Atom number (IATOM), number of neighbours including the atom itself (NNEIGH), number of orbitals on the atom (NORB)
- For every neighbour of every atom:
  - Atom number (IATOM1), neighbour number (INEIGH), corresponding image atom to the neighbour in the central cell (IATOM2F), coefficients of the translation vector between the neighbour and its corresponding image (ICELL(1), ICELL(2), ICELL(3)). Between the coordinates of the neighbour  $\mathbf{r}_{\text{INEIGH}}$  and the image atom  $\mathbf{r}_{\text{IATOM2F}}$  the relation
 
$$\mathbf{r}_{\text{INEIGH}} = \mathbf{r}_{\text{IATOM2F}} + \sum_{i=1}^3 \text{ICELL}(i) \mathbf{a}_i$$
 holds, where  $\mathbf{a}_i$  are the lattice vectors of the supercell.
  - The corresponding part of the sparse matrix. The data block consists of NORB(IAT1) lines and NORB(IAT2F) columns.

The files are produced if requested by WriteRealHS = Yes (see section 2.5).

### 4.3 eigenvec.out, eigenvec.bin

These files contain the eigenvectors from the Hamiltonian, stored either as plain text (eigenvec.out) or in the native binary format of your system (eigenvec.bin).

The plain text format file eigenvec.out contains a list of the values of the components of each eigenvector for the basis functions of each atom. The atom number in the geometry, its chemical type and the particular basis function are listed, followed by the relevant value from the current eigenvector and then the Mulliken population for that basis function for that level. The particular eigenvector,  $k$ -point and spin channel are listed at the start of each set of eigenvector data. In the case of non-collinear spin, the format is generalised for spinor wavefunctions. Complex coefficients for both the up and down parts of the spinors are given (instead of single eigenvector coefficient) followed by four values – total charge, then  $(x, y, z)$  magnetisation.

The binary format file eigenvec.bin contains the (unique) runId of the DFTB<sup>+</sup> simulation which produced the output followed by the values of the eigenvectors. The eigenvector data is ordered so that the individual components of the current eigenvector are stored, with subsequent eigenvectors for that  $k$ -point following sequentially. All  $k$ -points for the current spin channel are printed in this order, followed by the data for a second channel if spin polarised.

The files are produced if requested by setting `WriteEigenvectors = Yes`, with `EigenvectorsAsTxt` being also required to produce the plain text file (see section 2.6 for details).

## 4.4 charges.bin

The file `charges.bin` contains the orbitally-resolved charges for each atom, ordered as the charges on each orbital of an atom for a given spin channel, then each spin channel and finally over each atom. In later versions of DFTB<sup>+</sup> this format includes a check sum for the total charge and magnetisation. In the case of orbital potentials (p. 41) the file also contains extra population information for the occupation matrices.

This file is produced as part of the mechanism to restart SCC calculations, see sections 2.5 and 2.3.5.

## 4.5 md.out

This file is only produced for `VelocityVerlet{}` calculations (See p. 18). It contains a log of information generated during MD calculations, and appended every `MDRestartFrequency` steps. In the case of small numbers of atoms and long MD simulations it may be useful to set `WriteDetailedOut` to `No` and examine the information stored in this file instead.

## 4.6 Excited state results files

Several files are produced during excited state calculations depending on the particular settings from section 2.7.

**Note:** in the case of degeneracies, the oscillator strengths depend on arbitrary phase choices made by the ground state eigensolver. Only the sum over the degenerate contributions is well defined for most single particle transition properties, and label ordering of states may change if changing eigensolver or platform. For the excited state, properties like the intensities for individual excitations in degenerate manifolds again depend on phase choices made by both the ground and excited eigensolvers.

### 4.6.1 ARPACK.DAT

Internal details of the ARPACK solution vectors, see the ARPACK documentation [33] for details.

### 4.6.2 COEF.DAT

Data on the projection of this specific excited state onto the ground state orbitals. For the specific excited state, the (complex) decomposition of its single particle states onto the ground state single particle levels, together with its fractional contribution to the full excited state are given.

General format:

TF	Legacy flags
1 1.9999926523 2.0000000000	level 1, fraction of total WF, 2.0
-0.1944475716 0.0000000000 -0.1196876988 0.0000000000 ....	real then imaginary projection of level 1
	onto ground state 1, then ground state 2, etc.
-0.1196876988 0.0000000000 -0.1944475703 0.0000000000 ....	
.	
.	
.	
2 1.9999866161 2.0000000000	level 2
-0.2400145188 0.0000000000 -0.1767827333 0.0000000000 ....	real then imaginary projection of state 2
.	
.	
.	

4.6.3 EXC.DAT

Excitations data including the energies, oscillator strength, dominant Kohn-Sham transitions and the symmetry.

Example first few transitions for C<sub>4</sub>H<sub>4</sub>:

w [eV]	Osc.Str.	Transition	Weight	KS [eV]	Sym.
=====					
5.551	0.5143882	11 -> 12	1.000	4.207	S
5.592	0.0000000	10 -> 12	1.000	5.592	S

Two examples of singlet transitions with energies of 5.551 and 5.592 eV. The first is dipole allowed, the second not. In both cases they are transitions primarily (weight of 1.000) to single particle state 12, and are of singlet character (“S”).

In the case of spin-polarised calculations, an additional column of values are given instead of the symmetry, showing the level of spin contamination in the state (labelled as D<S\*S>), with typically states where a magnitude of less than 0.5 is usually considered reliable [38].

4.6.4 SPX.DAT

Single particle excitations (SPX) for transitions between filled and empty single particle states of the ground state. These are given in increasing single particle energy and show the oscillator strength and index of the Kohn-Sham-like states that are involved.

#	w [eV]	Osc.Str.	Transition
=====			



1	5.403	0.2337689	15	->	16
2	5.403	0.2337689	14	->	16
3	5.403	0.2337689	15	->	17
4	5.403	0.2337689	14	->	17
5	6.531	0.0000000	13	->	16
6	6.531	0.0000000	12	->	16

#### 4.6.5 TDP.DAT

Detail of the magnitude and direction of the transition dipole from the ground to excited states.

#### 4.6.6 TRA.DAT

Decomposition of the transition from the ground state to the excited states. The energy and spin symmetry are given together with the contributions from each of the single particle transitions.

#### 4.6.7 TEST\_ARPACK.DAT

Tests on the quality of the eigenvalues and vectors returned by ARPACK. For the  $i^{\text{th}}$  eigen-pair, the eigenvalue deviation corresponds to the deviation from  $(\langle \mathbf{x}_i | H | \mathbf{x}_i \rangle - \epsilon_i)$ , The eigen-vector deviation is a measure of rotation of the vector under the action of the matrix:  $|(H|\mathbf{x}_i\rangle - \epsilon_i|\mathbf{x}_i\rangle)|_2$ , the normalisation deviation is  $\langle \mathbf{x}_i | \mathbf{x}_i \rangle - 1$  and finally largest failure in orthogonality to other eigenvectors is given.

Example:

State	Ei deviation	Evec deviation	Norm deviation	Max non-orthog
1	-0.19428903E-15	0.80601119E-15	0.19984014E-14	0.95562226E-15
2	0.27755576E-16	0.85748374E-15	0.48849813E-14	0.36924443E-15
3	-0.12490009E-15	0.88607302E-15	0.88817842E-15	0.60384195E-15

#### 4.6.8 XCH.DAT

Net charges on atoms in the specified excited state. The top line contains the symmetry (Singlet or Triplet) and the number of the excited state. The next line is the number of atoms in the structure followed by some header text. Then on subsequent lines the number of each atom in the structure and its net charge are printed.

#### 4.6.9 XplusY.DAT

Expert file with the RPA  $(X + Y)_{ia}^{\Omega}$  data for all the calculated excited states.

Line 1: number of single particle excitations and the number of calculated excited states

Line 2: Level number 1, nature of the state (S, T, U or D) then excitation energy (in Hartree)

Line 3: expansion in the KS single particle transitions

.

.

.

Line 2: Level number 2, nature of the state (S, T, U or D) then excitation energy (in Hartree)

#### **4.6.10 XREST.DAT**

Net dipole moment of the specified excited state in units of Debye.

# Chapter 5

## MODES

The MODES program calculates vibrational modes using data created by DFTB<sup>+</sup>.

### 5.1 Input for MODES

The input file for MODES must be named `modes_in.hsd` and should be a Human-friendly Structured Data (HSD) formatted file (see Appendix A). The program can read the input in XML instead of HSD format if the input file is `modes_in.xml`. The input file must be present in the working directory. As with DFTB<sup>+</sup> to prevent ambiguity, the parser refuses to read in any input if two types of input file are present.

The table below contains the list of the properties, which must occur in the input file `modes_in.hsd`:

Name	Type	Condition	Default	Page
Geometry	plm		-	<a href="#">12</a>
Hessian	p		{}	<a href="#">84</a>
SlaterKosterFiles	plm		-	

Additionally optional definitions may be present:

Name	Type	Condition	Default	Page
DisplayModes	p		-	<a href="#">84</a>
Atoms	i+lm		1:-1	
WriteHSDInput	l		Yes	
WriteXMLInput	l		No	
RemoveTranslation	l		No	
RemoveRotation	l		No	

**Geometry** Specifies the geometry for the system to be calculated. See p. [12](#).

**Hessian** Contains the second derivatives matrix of the system energy with respect to atomic positions. See p. [84](#).

**SlaterKosterFiles** Name of the Slater-Koster files for every atom type pair combination. See p. [38](#).

**DisplayModes** Optional settings to plot the eigenmodes of the vibrations. See p. [84](#).

**Atoms** Optional list of atoms, ranges of atoms and/or the species of atoms for which the Hessian has been supplied. *This must be equivalent to the setting you used for MovedAtoms in your DFTB<sup>+</sup> input when generating the Hessian.*

**WriteHSDInput** Specifies, if the processed input should be written out in HSD format. (You shouldn't turn it off without good reason.)

**WriteXMLInput** Specifies, if the processed input should be written out in XML format.

**RemoveTranslation** Explicitly set the 3 translational modes of the system to be at 0 frequency.

**RemoveRotation** Explicitly set the rotation modes of the system to be at 0 frequency. Note, for periodic systems, this is usually incorrect (if used for a molecule full inside the central cell, it may be harmless).

### 5.1.1 Hessian{}

Contains the second derivatives of the energy supplied by DFTB<sup>+</sup>, see p. 17 for details of the options to generate this data. The derivatives matrix must be stored as the following order: For the  $i, j$  and  $k$  directions of atoms  $1 \dots n$  as

$$\frac{\partial^2 E}{\partial x_{i1} \partial x_{i1}} \frac{\partial^2 E}{\partial x_{j1} \partial x_{i1}} \frac{\partial^2 E}{\partial x_{k1} \partial x_{i1}} \frac{\partial^2 E}{\partial x_{i2} \partial x_{i1}} \frac{\partial^2 E}{\partial x_{j2} \partial x_{i1}} \frac{\partial^2 E}{\partial x_{k2} \partial x_{i1}} \dots \frac{\partial^2 E}{\partial x_{kn} \partial x_{kn}}$$

*Note:* for supercell calculations, the modes are currently obtained at the  $\mathbf{q} = 0$  point, irrespective of the k-point sampling used.

### 5.1.2 DisplayModes{}

Allows the eigenvectors of the system to be plotted out if present

PlotModes	i+lm	1:-1
Animate	1	Yes
XMakeMol	1	Yes

**PlotModes** Specifies list of which eigenmodes should be plotted as xyz files. Remember that there are  $3N$  modes for the system (including translation and rotation).

**Animate** Produce separate animation files for each mode or a single file multiple modes where the mode vectors are marked for each atom.

**XMakeMol** Adapt xyz format output for XMakeMol dialect xyz files.

## Chapter 6

# WAVEPLOT

The WAVEPLOT program is a tool for the visualisation of molecular orbitals. Based on the files created by a calculation performed by DFTB<sup>+</sup> it is capable of producing three dimensional information about the charge distribution. The information is stored as cube files, which can be visualised with many common graphical tools (e.g. VMD or JMOL).

The user controls WAVEPLOT through an input file, choosing which orbitals and charge distributions should be plotted for which spatial region. Since the information about the shape of the basis functions is usually not contained in the Slater-Koster files, the coefficients and exponents for the Slater type orbitals must be entered by the user as part of the input file.

The WAVEPLOT tool offers the following plotting capabilities:

- Total charge density.
- Total spin polarisation.
- Difference between the total charge density and the density obtained by the superposition of the neutral atomic densities (visualisation of the charge shift).
- Electron density for individual levels.
- Real and imaginary part of the wavefunctions for individual levels.

### 6.1 Input for WAVEPLOT

The input file for WAVEPLOT must be named `waveplot_in.hsd` and should be a Human-friendly Structured Data (HSD) formatted file (see Appendix A) The program can read the input in XML instead of HSD format if the input file is `waveplot_in.xml`. The input file must be present in the working directory. To prevent ambiguity, the parser refuses to read in any input if both files are present.

The table below contains the list of the properties, which must occur in the input file `waveplot_in.hsd`:

Name	Type	Condition	Default	Page
Options	p		-	<a href="#">86</a>
DetailedXML	s		-	
EigenvecBin	s		-	
GroundState	s		Yes	
Basis	p		-	<a href="#">90</a>

**Options** Contains the options for WAVEPLOT. See p. [86](#).

**DetailedXML** Specifies the name of the file, which contains the detailed XML output of the DFTB<sup>+</sup> calculation (presumably detailed.xml).

**EigenvecBin** Specifies the name of the file, which contains the eigenvectors in binary format (presumably eigenvec.bin).

**GroundState** Read ground or excited state occupation data from the detailed XML output.

**Basis** Contains the definition of the Slater-type orbitals which were used as basis in the DFTB<sup>+</sup> calculation. At the moment, due to technical reasons this information has to be entered by the user per hand. In a later stage, it will be presumably read in by WAVEPLOT automatically. See p. [90](#).

Additionally optional definitions may also be present:

Name	Type	Condition	Default	Page
ParserOptions	p		{ }	<a href="#">92</a>

### 6.1.1 Options

This property contains the options (as a list of properties), which the user can set, in order to influence the behaviour of WAVEPLOT. The following properties can be specified:

PlottedRegion	plm		-	<a href="#">88</a>
NrOfPoints	3i		-	
PlottedKPoints	i+lm	periodic system	-	
PlottedLevels	i+lm		-	
PlottedSpins	i+lm		-	
TotalChargeDensity	1		No	
TotalSpinPolarisation	1		No	
TotalChargeDifference	1		No	
TotalAtomicDensity	1		No	
ChargeDensity	1		No	
RealComponent	1		No	
ImagComponent	1	complex wavefunction	No	
FoldAtomsToUnitCell	1	periodic system	No	
FillBoxWithAtoms	1		No	
NrOfCachedGrids	i		-1	
Verbose	1		No	
RepeatBox	3i		{1 1 1}	
ShiftGrid	1		Yes	

**PlottedRegion** Regulates the region which should be plotted. See p. 88.

**NrOfPoints** Specifies the resolution of the equidistant grid on which the various quantities should be calculated. The three integers represent the number of points along the three vectors of the parallelepiped specifying the plotted region. The number of all calculated grid points is the product of the three integers.

Example:

```
NrOfPoints = { 50 50 50 }    # 125 000 grid points
```

**PlottedKPoints** The list of integers specified here represent the k-points, in which the molecular orbitals should be plotted. The first k-point in the original DFTB<sup>+</sup> calculation is represented by "1". The order of the specified k-points does not matter. You can also use the specification of the form from:to to specify ranges. (For more details on range specification, see the MovedAtoms keyword in the DFTB<sup>+</sup> manual.) The actual list of molecular orbitals to plot is obtained by intersecting the specifications for PlottedKPoints, PlottedLevels and PlottedSpins. The option is ignored if the original calculation was not periodic.

Example:

```
PlottedKPoints = 1 3 5      # The 1st, 3rd and 5th k-point is plotted
```

**PlottedLevels** The list of integers specified here represent the states, which should be plotted. The first (lowest lying) state in the original DFTB<sup>+</sup> calculation is represented by "1". The order of the specified states does not matter. You can also use the specification of the form from:to to specify ranges. (For more details on range specification, see the MovedAtoms keyword in the DFTB<sup>+</sup> manual.) The actual list of molecular orbitals to plot is obtained by intersecting the specifications for PlottedKPoints, PlottedLevels and PlottedSpins.

Example:

```
PlottedLevels = 1:-1      # All levels plotted
```

**PlottedSpins** The list of integers specified here represent the spins, for which the molecular orbitals should be plotted. The first spin in the original DFTB<sup>+</sup> calculation is represented by "1". The order of the specified spins does not matter. You can also use the specification of the form from:to to specify ranges. (For more details on range specification, see the MovedAtoms keyword in the DFTB<sup>+</sup> manual.) The actual list of molecular orbitals to plot is obtained by intersecting the specifications for PlottedKPoints, PlottedLevels and PlottedSpins.

Example:

```
PlottedSpins = 1 2        # Both spin-up and spin-down plotted
```

**ChargeDensity** If true, the absolute square of the wavefunction is plotted for the selected molecular orbitals.

**RealComponent** If true, the real component of the wavefunction is plotted for the selected molecular orbitals.

**ImagComponent** If true, the imaginary component of the wavefunction is plotted for the selected molecular orbitals. This option is only parsed, if the wavefunctions in the DFTB<sup>+</sup> calculation were complex.

**TotalChargeDensity** If true, the total charge density of the system is plotted.

**TotalSpinPolarisation** If true, the total spin polarisation of the system (difference of the spin up and spin down densities) is plotted. This option is only interpreted if the processed DFTB<sup>+</sup> calculation was spin polarised.

**TotalChargeDifference** If true, the difference between the total charge density and the charge density obtained by superposing the neutral atomic densities is plotted.

**TotalAtomicDensity** If true, the superposed neutral atomic densities are plotted.

**FoldAtomsToUnitCell** If true, the atoms are folded into the parallelepiped unit cell of the crystal.

**FillBoxWithAtoms** If true, the geometry is extended by those periodic images of the original atoms, which falls in the plotted region or on its borders. It sets FoldAtomsToUnitCell to Yes.

**NrOfCachedGrids** Specifies how many grids should be cached at the same time. The value -1 stands for as many as necessary to be as fast as possible. Since the plotted grids could eventually become quite big, you should set it to some positive non-zero value if you experience memory problems.

Example:

```
NrOfCachedGrids = 5    # Maximal 5 cached grids
```

**RepeatBox** The three integers specify how often the plotted region should be repeated in the generated cube files. Since repeating the grid is not connected with any extra calculations, this is a cheap way to visualise a big portion of a solid. You want probably set the FillBoxWithAtoms option to Yes to have the atoms also repeated (otherwise only the plotted function is repeated). In order to obtain the correct picture, you should set the plotted region to be an integer multiple of the unit cell of the crystal. Please note, that the phase of the wavefunctions in the repeated cells will be incorrect, except in the  $\Gamma$ -point.

Example:

```
RepeatBox = { 2 2 2 }    # Visualising a 2x2x2 supercell
```

**ShiftGrid** Whether the grid should be shifted, so that the specified origin lies in the middle of a cell and the grid fills out the specified plotted region symmetrically. The default is Yes. If set to No, the specified grid origin will be at the edge of a cell.

**Verbose** If true, some extra messages are printed out during the calculation.

### PlottedRegion

Specifies the region, which should be included in the plot. You can specify it explicitly (as property list), or let WAVEPLOT specify it automatically using either the `UnitCell{}` or the `OptimalCuboid{}` methods.



**Explicit specification** Specifies origin and box size explicitly.

Origin	3r	-
Box	9r	-

**Origin** [*length*] Specifies the xyz coordinates of the origin as three real values.

**Box** [*length*] Specifies the three vectors which span the parallelepiped of the plotted region. The vectors are specified sequentially ( $a_{1x}$   $a_{1y}$   $a_{1z}$   $a_{2x}$   $a_{2y}$   $a_{2z}$   $a_{3x}$   $a_{3y}$   $a_{3z}$ ). You are allowed to specify an arbitrary parallelepiped with nonzero volume here. Please note, however, that some visualisation tools only handles cube files with cuboid boxes correctly.

Example:

```
PlottedRegion = {
  Origin = { 0.0 0.0 0.0 }
  Box [Angstrom] = {
    12.5 12.5 -12.5
    12.5 -12.5 12.5
    -12.5 12.5 12.5
  }
}
```

**UnitCell{}** For the periodic geometries, this method specifies the plotted region to be spanned by the three lattice vectors of the crystal. The origin is set to (0 0 0). For cluster geometries, the smallest cuboid containing all atoms is constructed. For a cluster geometry the UnitCell{} object may have the following property:

MinEdgeLength	r	1.0
---------------	---	-----

**MinEdgeLength** [*length*] Minimal side length of the cuboid, representing the plotted region. This helps to avoid cuboids with vanishing edge lengths (as it would be the case for a linear molecule).

Example:

```
PlottedRegion = UnitCell {
  MinEdgeLength [Bohr] = 2.0
}
```

**OptimalCuboid{}** Specifies the plotted region as a cuboid, which contains all the atoms and enough space around them, that no wavefunctions are leaking out of the cuboid. This object does not have any parameters.

Example:

```
PlottedRegion = OptimalCuboid {}
```



```

AtomicNumber = 1
Orbital = {          # 1s orbital
  AngularMomentum = 0
  Occupation = 1
  Cutoff = 4.2
  Exponents = { 2.00000  1.00000 }
  Coefficients = {
    1.374518455977e+01  1.151435212242e+01  2.072671588012e+00
    -1.059020844305e+01  3.160957468828e+00 -2.382382105798e-01
  }
}
}
}
}

```

### Basis for an atom type

The actual basis for every atom type is specified as a property with the name of that type:

AtomicNumber	i	-	
Orbital	p	-	91
⋮			

**AtomicNumber** The atomic number of the species. This is not needed in the actual calculations, but for creating proper cube-files.

**Orbital** Contains the parameters of the orbitals. For every orbital a separate Orbital block must be created. See below.

**Orbital** For every orbital there is an orbital block which specifies the radial wavefunction. Thereby the following properties must be used:

AngularMomentum	i	-
Occupation	r	-
Cutoff	r	-
Exponents	r+	-
Coefficients	r+	-

**AngularMomentum** Angular momentum of the current orbital. ( $s - 0$ ,  $p - 1$ ,  $d - 2$ ,  $f - 3$ )

**Occupation** Occupation of the orbital in the neutral ground state. (Needed to obtain the right superposed atomic densities.)

**Cutoff** Cutoff for the wave function. You should choose a value, where the value of  $4\pi r^2 |R(r)|^2$  drops below  $10^{-4}$  or  $10^{-5}$ .  $R(r)$  is the radial part of the wave function. If you do not have the possibility to visualise the radial wave function, take the half of the longest distance, for which an overlap interaction exists in the appropriate homonuclear Slater-Koster file. (Value must be entered in Bohr.)

**Exponents** The radial wave function with angular momentum  $l$  has the form:

$$R_l(r) = \sum_{i=1}^{n_{\text{exp}}} \sum_{j=1}^{n_{\text{pow}}} c_{ij} r^{l+j-1} e^{-\alpha_i r} \quad (6.1)$$

This property defines the multiplication factors in the exponent ( $\alpha_i$ ).

**Coefficients** This property contains the coefficients  $c_{ij}$  as defined in equation (6.1). The sequence of the coefficients must be as follows:

$$c_{11} \ c_{12} \ \dots \ c_{1n_{\text{pow}}} \ c_{21} \ c_{22} \ \dots \ c_{2n_{\text{pow}}} \ \dots$$

### 6.1.3 ParserOptions

This block contains the options, which are effecting only the behaviour of the HSD/XML parser and are not passed to the main program.

IgnoreUnprocessedNodes	1	No
StopAfterParsing	1	No

**IgnoreUnprocessedNodes** By default the code stops if it detects unused or erroneous keywords in the input. This *dangerous* flag suspends these checks. Use only for debugging purposes.

**StopAfterParsing** If set to Yes, the parser stops after processing the input and written out the processed input to the disc. It can be used to make sanity checks on the input without starting an actual calculation.

# Appendix A

## The HSD format

The Human-friendly Structured Data (HSD) format is a structured input format, which can be bijectively mapped onto a subset of the XML-language. Its simplified structure and notation should make it a more convenient user interface than reading and writing XML tags. This section contains a brief overview of the most important aspects of this format.

An input file in the HSD format consists basically of property assignments of the form

Property = value

where the value value was assigned to the property Property. The value must be one of the following types (detailed description of each follows later on):

- Scalar, such as
  - integer
  - real
  - logical
  - string
- list of scalars
- method
- list of further property assignments

An unquoted hash mark (#) is interpreted as the start of a comment, everything after it, up to the end of the current line, is ignored by the parser (hash marks inside of quotes are taken as literals not comments):

# Entire line with comment

Prop1 = "hell#oo" # Note, that the first hashmark is quoted!

The name of the properties, the methods and the logical values are case insensitive, so the assignments

```

Prop1 = 12
prOP1 = 12
Prop2 = Yes
Prop2 = YES

```

are pairwise identical. Quoted strings (specified either as a value for a property or as a file name), however, are case sensitive.

Due to technical issues, the maximal line length is currently limited to 1024 characters. Lines longer than this are chopped without warning.

If a property, which should only appear once, is defined more than once, the parser uses the *first* definition and ignores all the other occurrences. *Thus specifying a property in the input a second time, does not override the first definition.* (For advanced use the HSD syntax also offers the possibility of conditional overriding/extending of previous definitions. For more details see [A.6.](#))

## A.1 Scalars and list of scalars

The following examples demonstrate the assignments with scalar types:

```

SomeInt = 1
SomeInt2 = -3
SomeRealFixedForm = 3.453
SomeRealExpForm = 2.12e-45
Logical1 = Yes
Logical2 = no
SomeString = "this is a string value"

```

As showed above, real numbers can be entered in either fixed or exponential form. The value for logical properties can be either Yes or No (case insensitive). Strings should always be enclosed in quotation marks, to make sure that they are treated as one string and that they are not interpreted by the parser:

```

String1 = "quoted string"
String2 = this value is actually a list of 9 strings # list of strings!
String3 = "Method { ;" # This is a string assignment
String4 = Method { # This is syntactically incorrect, since
                # it tries to assign a method to String4

```

A list of scalars is created by sequentially writing the scalars separated by one or more spaces:

```

PlottedLevels = 1 2 3
Origin = 0.0 0.0 0.0
ConfirmItTwice = Yes Yes
SpeciesNames = "Ga" "As"

```

The assignments statements are usually terminated by the end of the line. If the list of the assigned values goes over several lines, it must be entered between curly (brace) brackets. In that case, instead of the line end, the closing bracket will signal the end of the assignment. It is allowed to put a list of scalars in curly brackets, even if it is only one line long.

```

PlottedLevels = {
  1 2 3
}
Origin = { 0.0
0.0 0.0 }
Short = { 1 2 3 }

```

If you want to put more than one assignment in a line, you have to separate them with a semi-colon:

```
Variable = 12; Variable2 = 3.0
```

If a property should be defined as empty, either the empty list must be assigned to it or it must be defined as an empty assignment terminated by a semi-colon:

```

EmptyProperty = {}
EmptyProperty2 = ;

```

Please note, that explicitly specifying a property to be empty is not the same as not having specified it at all. In the latter case, the parser substitutes the default value for that property (if there is a default for it), while in the first case it interprets the property to be empty. If a property without default value is not specified, the parser stops with an appropriate error message.

## A.2 Methods and property lists

Besides the scalar values and the list of scalars, the right hand side of an assignment may also contain a method, which itself may contain one or more scalar values or further property assignments as parameters:

```

Diagonaliser = LapackDAC {}    # Method without further params
PlottedLevels = Range { 1 3 }  # Range needs two scalar params
PlottedRegion = UnitCell {    # UnitCell needs a property list
  MinEdgeLength = 1.0        # as parameter
  SomeOtherProperty = Yes
}

```

The first assignment above is an example, where the method on the right hand side does not need any parameters specified. Please note, that even if no parameters are required, the opening and closing brackets after the method are mandatory. If the brackets are missing, the parser interprets the value as a string.

In the second assignment, the method `Range` needs only two integers as parameters, while for the method `UnitCell` several properties must be specified. A method may contain either nothing or scalars or property assignments, but never scalars and property assignments together. So the following assignment would be invalid:

```

InvalidSpecif = SomeMethod {
  1 2 3
  Property1 = 12
  "Some strings here"
}

```

Very often a value for the property is represented by a list of further property assignments (as above, but without naming an explicit method beforehand). In that case, the property assignments must be put between curly brackets (property list):

```
Options = {
  SubOption1 = 12
  Suboption2 = "string"
}
```

### A.3 Modifiers

Each property may carry a modifier, which changes the interpretation of the assigned value:

```
LatticeConstant [Angstrom] = 12.23
```

Here, the property `LatticeConstant` possesses the `Angstrom` modifier, so the specified value will be interpreted to be in Ångström instead of the default length unit. Specifying a modifier for a property which is not allowed to carry one leads to parsing error.

The syntax of the HSD format also allows methods (used as values on the right hand side of an assignment) to carry modifiers, but this is usually not used in the current input structures.

Sometimes, the assigned value to a property contains several values with different units, so that more than one modifiers can be specified. In that case, the modifiers must be separated by a comma.

```
VolumeAndChargePerElement [Angstrom^3,au] = {
  1.2 0.3 # first element
  4.2 0.1 # second element
}
```

You have to specify either no modifier or all modifiers. If you want specify the default units for some of the quantities, you can omit the name of the appropriate modifier, but you must include the separating comma:

```
# Specifying the default unit for the charge. Note the separating comma!
VolumeAndChargePerElement [Angstrom^3,] = {
  1.2 0.3 # first element
  4.2 0.1 # second element
}
```

Specifying not enough or too many modifiers leads to parser error.

### A.4 File inclusion

It is possible to include files in an HSD-formatted input by using the `<<<` and `<<+` operators. The former includes the specified file as raw text without parsing it, while latter parses the included text:



```

Geometry = GenFormat {
  <<< "geo_start.gen"
}
Basis = {
  <<+ "File__containing__the__property__definitons__for__the__basis"
}

```

The file included with the <<+ operator must be a valid HSD document in itself.

## A.5 Processing

After having parsed and processed the input file, the parser writes out the processed input to a separate file in HSD format. This file contains the internal representation for all properties, which can be specified by the user. In particular, all default values are explicitly set and all automatic definitions (e.g. ranges) are converted to their internal representations.

Assuming the following example as input

```

# Lattice constant specified in Angstrom.
# Internal representation uses Bohr, so it will be converted.
LatticeConstant [Angstrom] = 12.0

# This property is not set, as its commented out, so the
# default value will be set for this (let's assume, it's Yes)
#DoAProperJob = No

# Plotted levels specified as a range with parameter 1:3.
# This will be replaced by an explicit listing of the levels
PlottedLevels = { 1:3 }

```

the parsed and processed input (written to a special file) should look something like

```

LatticeConstant = 22.676713499923075
DoAProperJob = Yes
PlottedLevels = {
  1 2 3
}

```

If you want to reproduce your calculation later, you should use this processed input. It should give you identical results, even if the default setting for some properties had been changed in the code.

Since the HSD format is mapped by the parser internally to an XML tree, most codes using this format allow (or hopefully will allow) to dumping out of the processed input in the XML format as well, and to use that later as an input, instead of the HSD formatted input. This is practical for people preferring to work with XML or if the input should be automatically generated by a script.

## A.6 Extended format

As stated earlier, if a property, which should be defined only once, occurs more than once in the input, the parser uses per default the first definition and ignores all the others. Sometimes this is not

the desired behaviour, therefore, the HSD format also offers the possibility to override properties that were set earlier. This feature can be very useful for scripts which generate HSD input based on some user provided template. By just appending a few lines to the end of the user provided input the scripts can make sure that certain properties are set correctly. Thus, the script can modify the user input, without having to parse it at all.

The parser builds internally an XML DOM-tree from the HSD input. For every property or method name an XML tag with the same name (but lowercased) is created, which will contain the value of the property or the method. If the value contains further properties or methods, new XML tags are created inside the original one. Shortly, the HSD input is mapped on a tree, whereas the assignment and the containment (equal sign and curly brace) are turned into a parent-child relationships.<sup>1</sup> As an example an HSD input and the corresponding XML-representation is given below:

Level0Elem1 = 1	<level0elem1>1</level0elem1>
Level0Elem2 = { 1 2 3 }	<level0elem2>1 2 3</level0elem2>
Level0Elem3 = {	<level0elem3>
Level1Elem1 = 12	<level1elem1>12</level1elem1>
Level1Elem2 = Level2Elem1 {	<level1elem2>
	<level2elem1>
Level3Elem1 = "abcd"	<level3elem1>"abcd"</level3elem1>
Level3Elem2 = {	<level3elem2>
Level4Elem1 = 12	<level4elem1>12</level4elem1>
}	</level3elem2>
}	</level2elem1>
}	</level1elem2>
	</level0elem3>

By prefixing property and method names, the default behaviour of the parser can be overridden. Instead of creating a new tag (on the current encapsulation level) with the appropriate name, it will look for the *first occurrence* of the given tag and will process that one. Depending of the prefix character, the tag is processed in the following ways:

- +:** If the tag exists already, it's value is modified, otherwise the parser stops.
- ?:** If the tag exists already, it's value is modified, otherwise the parser ignores the prefixed HSD construct.
- \*:** If the tag exists already, it's value is modified, otherwise it is created (and then it's value is modified).
- /:** If the tag does not exist yet, it is created and modified, otherwise the prefixed HSD construct is ignored.
- !:** The tag is newly created and modified. If it exists already, the old occurrence is deleted first.

The way the value of the tag is going to be modified, is ruled by the constructs inside the prefixed property or method name. If the parser finds non prefixed constructs here, the appropriate tags are just added, otherwise the behaviour is determined by the rules above, just acting one level deeper in the tree. The following examples should make this a little bit more clear.

<sup>1</sup>In the internal tree representation of the HSD input there is no difference between properties and methods, both are just elements capable to contain some value or further elements. The differentiation in the HSD input is artificial and is only for human readability (equal sign after property names, curly brace after method names),

- Changing the value of Level0Elem1 to 3. If the element does not exist, it should be created with the value 3.

```
!Level0Elem1 = 3
```

- Changing the value of Level0Elem3/Level1Elem1 to 21 (the slash indicates the parent-child relationship). If the element does not exist, stop with an error message:

```
# Make sure the containing element exists. If yes, go inside, otherwise die.
+Level0Elem3 = {
  # Set the value of Level1Elem1 or die, if it does not exist.
  +Level1Elem1 = 21
}
```

Please note, that each tag in the path must be prefixed. Using the following construct instead of the original one

```
# Not prefixed, so it creates a new tag with empty value
Level0Elem3 = {
  # The new tag doesn't contain anything, so the parser stops here
  +Level1Elem1 = 21
}
```

would end with an error message. Since Level0Elem1 is not prefixed here, a tag is created for it with an empty value (no children). It does not matter, whether the tag already existed before or not, a new tag is created and appended as the last element (last child) to the current block. Then the parser is processing its value. Due to the +Level1Elem1 directive it is looking for a child tag <level1elem1>. Since the tag was newly created, it does not contain any children, so the parser stops with an error message.

- Create a new tag Level1Elem3 inside Level0Elem3 with some special value. If the tag already exists, replace it.

```
# Modifying the children of Level0Elem3 or dying if not present
+Level0Elem3 = {
  # Replacing or if not existent creating Level1Elem3
  !Level1Elem3 = NewBlock {
    NewValue1 = 12
  }
}
```

This example also shows, that the value for the new property can be any arbitrary complex HSD construct.

- Provide a default value "string" for Level0Elem3/Level1Elem2/Level2Elem1/Level3Elem1. If the tag is already present do not change its value.

```
# Modify Level0Elem3 or create it if non-existent
*Level0Elem3 = {
  # Modify Level1Elem2 and Level2Elem1 or create them if non-existent
  *Level1Elem2 = *Level2Elem1 {
    # Create Level3Elem1 if non-existent with special value.
    /Level3Elem1 = "string"
  }
}
```

- If Level0Elem3/Level1Elem2 has the value Level2Elem1, make sure that Level3Elem1 in it exists, and has "" as value. If Level1Elem2 has a different value, do not change anything.

```
# If Level0Elem3 is present, process it, otherwise skip this block
?Level0Elem3 = {
  # The same for the next two containers
  ?Level1Elem2 = ?Level2Elem1 {
    # Create or replace Level3Elem1
    !Level3Elem1 = ""
  }
}
```

## Appendix B

### Unit modifiers

The DFTB<sup>+</sup> code uses internally atomic units (with Hartree as the energy unit). The value of every numerical property in the input is interpreted to be in atomic units (au), unless the property carries a modifier.

The allowed modifiers and the corresponding conversion factors are given below.<sup>1</sup> (The modifiers are case insensitive).

#### Length:

Angstrom, AA (for Ångström)	0.188972598857892E+01
Meter, m	0.188972598857892E+11
pm	0.188972598857892E-01
Bohr, au	1.000000000000000E+00

#### Mass:

amu	0.182288848492937E+04
au	1.000000000000000E+00
da	0.182288848492937E+04
dalton	0.182288848492937E+04

#### Volume:

Angstrom <sup>3</sup> , AA <sup>3</sup>	0.674833303710415E+01
meter <sup>3</sup> , m <sup>3</sup>	0.674833303710415E+31
pm <sup>3</sup>	0.674833303710415E-05
bohr <sup>3</sup> , au	1.000000000000000E+00

#### Energy:

Rydberg, Ry	0.500000000000000E+00
Electronvolt, eV	0.367493245336341E-01
kcal/mol	0.159466838598749E-02
Kelvin, K	0.316681534524639E-05
cm <sup>-1</sup>	0.455633507361033E-05
Joule, J	0.229371256497309E+18
Hartree, Ha, au	1.000000000000000E+00

---

<sup>1</sup>The conversion factors listed here were calculated with double precision on i686-linux architecture. Depending on your architecture, the values used there may deviate slightly.

**Force:**

eV/Angstrom, eV/Å	0.194469064593167E-01
Joule/meter, J/m	0.121378050512919E+08
Hartree/Bohr, Ha/Bohr, au	1.000000000000000E+00

**Time:**

femtosecond, fs	0.413413733365614E+02
picosecond, ps	0.413413733365614E+05
second, s	0.413413733365614E+17
au	1.000000000000000E+00

**Charge:**

Coulomb, C	0.624150947960772E+19
au, e	1.000000000000000E+00

**Velocity:**

au	1.000000000000000E+00
m/s	0.457102857516272E-06
pm/fs	0.457102857516272E-03
Å/ps	0.457102857516272E-04

**Pressure:**

Pa	0.339893208050290E-13
au	1.000000000000000E+00

**Frequency:**

Hz	0.241888432650500E-16
THz	0.241888432650500E-04
cm <sup>-1</sup>	0.725163330219952E-06
au	1.000000000000000E+00

**Electric field strength:**

v/m	0.194469063788953E-11
au	1.000000000000000E+00

**Dipole moment:**

CoulombMeter, Cm	0.117947426715764E+30
Debye	0.393430238326893E+00
au	1.000000000000000E+00

## Appendix C

# Description of the gen format

The general (gen) format can be used to describe clusters and supercells. It is based on the xyz format introduced with xmol, and extended to periodic structures. Unlike some earlier implementations of gen, the format should not include any neighbour mapping information.

The first line of the file contains the number of atoms,  $n$ , followed by the type of geometry. C for cluster (non-periodic), S for supercell in Cartesian coordinates or F for supercell in fractions of the lattice vectors. The supercells are periodic in 3 dimensions.

The second line contains the chemical symbols of the elements present separated by one or more spaces. The following  $n$  lines contain a list of the atoms. The first number is the atom number in the structure (this is currently ignored by the program). The second number is the chemical type from the list of symbols on line 2. Then follow the coordinates. For S and C format, these are  $x, y, z$  in Å, but for F they are fractions of the three lattice vectors.

If the structure is a supercell, the next line after the atomic coordinates contains the coordinate origin in Å (this is ignored by the parser). The last three lines are the supercell vectors in Å. These four lines are not present for clusters.

Example: Geometry of GaAs with 2 atoms in the fractional supercell format

```
2 F
Ga As
1 1 0.0 0.0 0.0
2 2 0.25 0.25 0.25
0.000000 0.000000 0.000000
2.713546 2.713546 0.
0. 2.713546 2.713546
2.713546 0. 2.713546
```





## Appendix D

### Atomic spin constants

These are suggested values for some atomic spin constants ( $W$  values) as given in reference [39], only the first two decimal places of the finite spin constants are numerically significant. These constants may eventually be included in the Slater-Koster files directly. Check the documentation of the Slater-Koster files required for a calculation to decide whether to use the LDA or PBE-GGA spin constants.

W		LDA			PBE		
		$s$	$p$	$d$	$s$	$p$	$d$
H	$s$	-0.064			-0.072		
C	$s$	-0.028	-0.024		-0.031	-0.025	
	$p$	-0.024	-0.022		-0.025	-0.023	
N	$s$	-0.030	-0.026		-0.033	-0.027	
	$p$	-0.026	-0.025		-0.027	-0.026	
O	$s$	-0.032	-0.028		-0.035	-0.030	
	$p$	-0.028	-0.027		-0.030	-0.028	
Si	$s$	-0.018	-0.013	0.000	-0.020	-0.015	0.000
	$p$	-0.013	-0.012	0.000	-0.015	-0.014	0.000
	$d$	0.000	0.000	-0.019	0.002	0.002	-0.032
S	$s$	-0.019	-0.016	0.000	-0.021	-0.017	0.000
	$p$	-0.016	-0.014	0.000	-0.017	-0.016	0.000
	$d$	0.000	0.000	-0.010	0.000	0.000	-0.080
Fe ( $3d^7 4s^1$ )	$s$	-0.013	-0.009	-0.003	-0.016	-0.012	-0.003
	$p$	-0.009	-0.011	-0.001	-0.012	-0.029	-0.001
	$d$	-0.003	-0.001	-0.015	-0.003	-0.001	-0.015
Ni	$s$	-0.009	-0.009	-0.003	-0.016	-0.012	-0.003
	$p$	-0.009	-0.010	-0.001	-0.012	-0.022	-0.001
	$d$	-0.003	-0.001	-0.017	-0.003	-0.001	-0.018



## Appendix E

### Dispersion constants

The following table contains recommended dispersion constants for some elements with the Slater-Kirkwood dispersion model (see p. 44). The values have been tested for biological systems, C, N, O and H predominantly for DNA [22]. If you would like to calculate different systems or you're looking for other elements, check references [40] and [41]. The values of the atomic polarisabilities and cutoffs are given for zero, one, two, three, four and more than four neighbours.

Element	Polarisability [ $\text{\AA}^3$ ]	Cutoff [ $\text{\AA}$ ]	Chrg	Note
O	0.560 0.560 0.000 0.000 0.000 0.000	3.8 3.8 3.8 3.8 3.8 3.8	3.15	PO <sub>4</sub> only S, not SO <sub>2</sub>
N	1.030 1.030 1.090 1.090 1.090 1.090	3.8 3.8 3.8 3.8 3.8 3.8	2.82	
C	1.382 1.382 1.382 1.064 1.064 1.064	3.8 3.8 3.8 3.8 3.8 3.8	2.50	
H	0.386 0.386 0.000 0.000 0.000 0.000	3.5 3.5 3.5 3.5 3.5 3.5	0.80	
P	1.600 1.600 1.600 1.600 1.600 1.600	4.7 4.7 4.7 4.7 4.7 4.7	4.50	
S	3.000 3.000 3.000 3.000 3.000 3.000	4.7 4.7 4.7 4.7 4.7 4.7	4.80	



## Appendix F

### Publications to cite

The following publications should be considered for citation, if you are publishing any results calculated by using DFTB<sup>+</sup>.

DFTB <sup>+</sup> code	[2]
non-SCC DFTB	[42], [43]
SCC DFTB	[30]
Collinear spin polarisation	[44]
Non-collinear spin polarisation	[45]
Spin orbit coupling	[45]
QM/MM coupling (external charges)	[46], [47]
Van der Waals interaction (dispersion)	[22]
DFTB+U	[19]
3rd order corrections	[29]
linear-response TD-DFTB	[32]



# Bibliography

- [1] D. A. Ryndyk, “DFTB<sup>+</sup> XT open software package for quantum nanoscale modeling,” <http://quantranspro.org/dftb+xt/>. 7
- [2] B. Aradi, B. Hourahine, and T. Frauenheim, “DFTB+, a Sparse Matrix-Based Implementation of the DFTB Method,” *J. Phys. Chem. A* **111**, 5678 (2007), <http://dftbplus.org/>. 7, 78, 109
- [3] A. Pecchia, G. Penazzi, L. Salvucci, and A. D. Carlo, “Non-equilibrium Green’s functions in density functional tight binding: method and applications,” *New Journal of Physics* **10**, 065022 (2008). 7, 61, 66
- [4] T. Frauenheim, G. Seifert, M. Elstner, T. Niehaus, C. Kohler, M. Amkreutz, M. Sternberg, Z. Hajnal, A. Di Carlo, and S. Suhai, “Atomistic simulations of complex materials: ground-state and excited-state properties,” *J. Phys. Cond. Matter* **14**, 3015–3047 (2002). 9
- [5] A. Kovalenko, S. Ten-no, and F. Hirata, “Solution of three-dimensional reference interaction site model and hypernetted chain equations for simple point charge water by modified method of direct inversion in iterative subspace,” *J. Comp. Chem.* **20**, 928–936 (1999). 17
- [6] H. C. Andersen, “Molecular dynamics at constant pressure and/or temperature,” *J. Chem. Phys.* **72**, 2384 (1980). 19
- [7] H. J. C. Berendsen, J. P. M. Postma, W. F. Van Gunsteren, A. Dinola, and J. R. Haak, “Molecular-Dynamics with Coupling to an External Bath,” *J. Chem. Phys.* **81**, 3684–3690 (1984). 19, 21
- [8] S. C. Harvey, R. K. Z. Tan, and T. E. Cheatham, “The flying ice cube: Velocity rescaling in molecular dynamics leads to violation of energy equipartition,” *J. Comp. Chem.* **19**, 726–740 (1998). 19
- [9] G. J. Martyna, M. E. Tuckerman, D. J. Tobias, and M. L. Klein, “Explicit reversible integrators for extended systems dynamics,” *Molecular Phys.* **87**, 1117–1157 (1996). 20
- [10] B. Aradi, A. M. N. Niklasson, and T. Frauenheim, “Extended Lagrangian Density Functional Tight-Binding Molecular Dynamics for Molecules and Solids,” *J. Chem. Theory Comput.* **11**, 3357–3363 (2015). 22, 51
- [11] M. Ceriotti, J. More, and D. E. Manolopoulos, “i-PI: A Python interface for ab initio path integral molecular dynamics simulations,” *Computer Phys. Comm.* **185**, 1019–1026 (2014). 24
- [12] D. D. Johnson, “Modified Broyden’s method for accelerating convergence in self consistent calculations,” *Phys. Rev. B* **38**, 12807 (2003). 30

- [13] V. Eyert, "A Comparative Study on Methods for Convergence Acceleration of Iterative Vector Sequences," *J. Comp. Phys.* **124**, 271 (1996). [31](#)
- [14] M. J. Han, T. Ozaki, and J. Yu, " $O(N)$  LDA+ $U$  electronic structure calculation method based on the nonorthogonal pseudoatomic orbital basis," *Phys. Rev. B* **73**, 045110 (2006). [36](#)
- [15] E. Anderson *et al.*, *LAPACK Users' Guide*, 3rd ed. (Society for Industrial and Applied Mathematics, Philadelphia, PA, 1999). [36](#)
- [16] M. Methfessel and A. T. Paxton, "High-precision sampling for Brillouin-zone integration in metals," *Phys. Rev. B* **40**, 3616 (1989). [37](#)
- [17] H. J. Monkhorst and J. D. Pack, "Special points for Brillouin-zone integrations," *Phys. Rev. B* **13**, 5188 (1976). [39](#)
- [18] H. J. Monkhorst and J. D. Pack, "'Special points for Brillouin-zone integrations'—a reply," *Phys. Rev. B* **16**, 1748 (1977). [40](#)
- [19] B. Hourahine, S. Sanna, B. Aradi, C. Köhler, T. Niehaus, and T. Frauenheim, "Self-Interaction and Strong Correlation in DFTB," *J. Phys. Chem. A* **111**, 5671 (2007). [41](#), [109](#)
- [20] A. G. Petukhov, I. I. Mazin, L. Chioncel, and A. I. Lichtenstein, "Correlated metals and the LDA+ $U$  method," *Phys. Rev. B* **67**, 153106–4 (2003). [41](#)
- [21] A. K. Rappe, C. J. Casewit, K. S. Colwell, W. A. Goddard III, and W. M. Skiff, "UFF, a full periodic table force field for molecular mechanics and molecular dynamics simulations," *J. Am. Chem. Soc.* **114**, 10024–10035 (1992). [43](#)
- [22] M. Elstner, P. Hobza, T. Frauenheim, S. Suhai, and E. Kaxiras, "Hydrogen bonding and stacking interactions of nucleic acid base pairs: a density-functional-theory based treatment," *J. Chem. Phys.* **114**, 5149 (2001). [43](#), [44](#), [45](#), [107](#), [109](#)
- [23] S. Grimme, J. Antony, S. Ehrlich, and H. Krieg, "A consistent and accurate ab initio parametrization of density functional dispersion correction (DFT-D) for the 94 elements H–Pu," *J. Chem. Phys.* **132**, 154104 (2010). [43](#), [45](#)
- [24] S. Grimme, S. Ehrlich, and L. Goerigk, "Effect of the Damping Function in Dispersion Corrected Density Functional Theory," *J. Chem. Phys.* **32**, 1456–1465 (2011). [43](#), [45](#)
- [25] L. Zhechkov, T. Heine, S. Patchkovskii, G. Seifert, and H. A. Duarte, "An Efficient a Posteriori Treatment for Dispersion Interaction in Density-Functional-Based Tight Binding," *J. of Chem. Theory Comput.* **1**, 841–847 (2005). [43](#)
- [26] J. Řezáč, "Empirical Self-Consistent Correction for the Description of Hydrogen Bonds in DFTB3," *J. Chem. Theory Comput.* **13**, 4804–4817 (2017). [47](#), [48](#), [49](#), [50](#)
- [27] J. Řezáč and P. Hobza, "Advanced Corrections of Hydrogen Bonding and Dispersion for Semiempirical Quantum Mechanical Methods," *J. Chem. Theory Comput.* **8**, 141–151 (2012). [47](#), [49](#), [50](#)
- [28] M. Gaus, Q. Cui, and M. Elstner, "DFTB3: Extension of the Self-Consistent-Charge Density-Functional Tight-Binding Method (SCC-DFTB)," *J. of Chem. Theory Comput.* **7**, 931–948 (2011). [47](#), [48](#)



- [29] Y. Yang, H. Yu, D. York, Q. Cui, and M. Elstner, "Extension of the Self-Consistent-Charge Density-Functional Tight-Binding Method: Third-Order Expansion of the Density Functional Theory Total Energy and Introduction of a Modified Effective Coulomb Interaction," *J. Phys. Chem. A* **111**, 10861 (2007). 47, 48, 109
- [30] M. Elstner, D. Porezag, G. Jungnickel, J. Elsner, M. Haugk, T. Frauenheim, S. Suhai, and G. Seifert, "Self-consistent-charge density-functional tight-binding method for simulations of complex materials properties," *Phys. Rev. B* **58**, 7260 (1998). 51, 109
- [31] J. Pipek and P. G. Mezey, "A fast intrinsic localization procedure applicable for *ab initio* and semiempirical linear combination of atomic orbital wave functions," *J. Chem. Phys.* **90**, 4916 (1989). 54
- [32] T. A. Niehaus, S. Suhai, F. Della Sala, P. Lugli, M. Elstner, G. Seifert, and T. Frauenheim, "Tight-binding approach to time-dependent density-functional response theory," *Phys. Rev. B* **63**, 085108 (2001). 55, 109
- [33] R. B. Lehoucq, D. C. Sorensen, and C. Yang, "ARPACK Users Guide: Solution of Large Scale Eigenvalue Problems by Implicitly Restarted Arnoldi Methods.," 1997. 55, 79
- [34] <https://github.com/opencollab/arpack-ng>. 55
- [35] D. Heringer, T. A. Niehaus, M. Wanko, and T. Frauenheim, "Analytical excited state forces for the time-dependent density-functional tight-binding method.," *J. Comp. Chem.* **28**, 2589 (2007). 56
- [36] A. Pecchia and A. D. Carlo, "Atomistic theory of transport in organic and inorganic nanostructures," *Reports on Progress in Physics* **67**, 1497 (2004). 66
- [37] A. Di Carlo, A. Pecchia, L. Latessa, T. Frauenheim, and G. Seifert, in *Introducing Molecular Electronics*, G. Cuniberti, K. Richter, and G. Fagas, eds., (Springer Berlin Heidelberg, Berlin, Heidelberg, 2005), pp. 153–184. 66
- [38] A. D. Garcia, Ph.D. thesis, Universität Bremen, 2014, <http://elib.suub.uni-bremen.de/edocs/00103868-1.pdf>. 80
- [39] C. Köhler, Ph.D. thesis, Department Physik der Fakultät für Naturwissenschaften an der Universität Paderborn, 2004, <http://ubdata.uni-paderborn.de/ediss/06/2004/koehler/>. 105
- [40] K. J. Miller, "Additivity methods in molecular polarizability," *J. Am. Chem. Soc.* **112**, 8533 (1990). 107
- [41] Y. K. Kang and M. S. Jhon, "Additivity of atomic static polarizabilities and dispersion coefficients," *Theoretica Chimica Acta* **61**, 41 (1982). 107
- [42] D. Porezag, T. Frauenheim, T. Köhler, G. Seifert, and R. Kaschner, "Construction of tight-binding-like potentials on the basis of density-functional theory: Application to carbon," *Phys. Rev. B* **51**, 12947 (1995). 109
- [43] G. Seifert, D. Porezag, and T. Frauenheim, "Calculations of molecules, clusters, and solids with a simplified LCAO-DFT-LDA scheme," *Int. J. Quant. Chem.* **58**, 185 (1996). 109
- [44] C. Köhler, G. Seifert, and T. Frauenheim, "Density-Functional based calculations for Fe(*n*), (*n* ≤ 32)," *Chem. Phys.* **309**, 23 (2005). 109

- [45] C. Köhler, T. Frauenheim, B. Hourahine, G. Seifert, and M. Sternberg, "Treatment of Collinear and Noncollinear Electron Spin within an Approximate Density Functional Based Method," *J. Phys. Chem. A* **111**, 5622 (2007). [109](#)
- [46] Q. Cui, M. Elstner, T. Frauenheim, E. Kaxiras, and M. Karplus, "Combined self-consistent charge density functional tight-binding (SCC-DFTB) and CHARMM," *J. Phys. Chem. B* **105**, 569 (2001). [109](#)
- [47] W. Han, M. Elstner, K. J. Jalkanen, T. Frauenheim, and S. Suhai, "Hybrid SCC-DFTB/Molecular Mechanical Studies of H-Bonded Systems and of N-acetyl-(L-Ala)<sub>n</sub>-N'-Methylamide Helices in Water Solution," *Int. J. Quant. Chem.* **78**, 459 (2000). [109](#)

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