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## Mechanistic Understanding of Oxygen Electrodes in Rechargeable Multivalent Metal-Oxygen Batteries

Zhuojian Liang<sup>[a]</sup> and Yi-Chun Lu<sup>\*[a]</sup>

Rechargeable multivalent metal-O<sub>2</sub> batteries (Mg-O<sub>2</sub>, Zn-O<sub>2</sub> and Al-O<sub>2</sub> etc.) are promising next-generation battery technologies for electric automobiles and large-scale energy storage owing to their high energy density, low cost, and sustainability. To realize the full potentials of these systems, an in-depth mechanistic understanding of the underlying chemistries is crucial for rational development. However, compared with the intensively studied Li-O<sub>2</sub> batteries, reaction mechanisms of

multivalent metal-O<sub>2</sub> batteries are much less understood. Here, we discuss the current status of mechanistic understanding of oxygen reduction and evolution reactions in rechargeable multivalent metal-O<sub>2</sub> batteries. We discuss the key open questions and future research directions of multivalent metal-O<sub>2</sub> batteries using examples and experiences from Li-O<sub>2</sub> batteries, which could guide and inspire the development of multivalent metal-O<sub>2</sub> batteries.

## 1. Introduction

High-energy-density, low-cost, and sustainable rechargeable batteries are critical foundations for wide market penetration of electric vehicles and stationary energy storage. Although current Li-ion batteries have achieved great success in these applications, they are limited by their energy density (200 Wh/kg).<sup>[1]</sup> The research community has been actively exploring new chemistries to advance towards high-energy batteries. Among the diverse systems, rechargeable metal-O<sub>2</sub> batteries are promising in meeting the energy requirements.

Metal-O<sub>2</sub> batteries combine a metal anode and a porous cathode in which O<sub>2</sub> acts as the active material to provide superior energy density compared to Li-ion batteries. Theoretically, metal-O<sub>2</sub> batteries promise specific energies up to 5,220 Wh/kg (Li-O<sub>2</sub> batteries based on Li<sub>2</sub>O) and energy densities up to 17,027 Wh/L (Al-O<sub>2</sub> batteries based on Al<sub>2</sub>O<sub>3</sub>). Practically, metal-O<sub>2</sub> batteries could achieve 3–5-fold higher specific energy than current Li-ion batteries.<sup>[1]</sup> In addition, without expensive elements such as Co in their active materials, the material cost of these batteries can be drastically lower than state-of-the-art Li-ion batteries. The stability of some multivalent metals in air, e.g. Al, Mg, Fe and Zn, could substantially improve the safety of rechargeable batteries over the Li/Na/K metal batteries. Contrary to the reactivity of alkali metal oxides (e.g. Li<sub>2</sub>O<sub>2</sub>, Na<sub>2</sub>O<sub>2</sub>) with air components (CO<sub>2</sub> and H<sub>2</sub>O), which requires the use of pure O<sub>2</sub> or air purification system, the stability of these multivalent metals oxides in the

air could allow the direct use of atmospheric air, which further reduces the cost and simplify the battery structure.

The development of rechargeable multivalent metal-O<sub>2</sub> batteries is still in its infancy. Most of the multivalent metal-O<sub>2</sub> batteries are limited by poor cycle life (the best being 320 cycles of Zn-O<sub>2</sub> batteries,<sup>[2]</sup> while less than 5 cycles in some other systems) and high energy loss. Improving the cycling stability and energy efficiency is strongly limited by the lack of understanding of the reaction mechanisms of the oxygen reduction reaction (ORR) and oxygen evolution reaction (OER). Resolving mechanistic open questions in reaction mechanism and degradation mechanism is one of the most crucial tasks in developing rechargeable multivalent metal-O<sub>2</sub> batteries.

Here, we first review the current status of mechanistic studies on oxygen electrodes in rechargeable multivalent metal-O<sub>2</sub> batteries. We then discuss future research directions of multivalent metal-O<sub>2</sub> batteries using examples and experiences learning from Li-O<sub>2</sub> batteries, which could guide and inspire the development of multivalent metal-O<sub>2</sub> batteries. Key open questions in the ORR and OER mechanism are identified.

## 2. Current Status of Mechanistic Understanding

2.1. Zn-O<sub>2</sub> Batteries

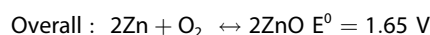
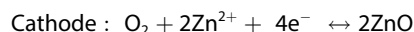
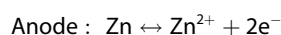
Most studies on Zn-O<sub>2</sub> batteries exploit alkaline aqueous electrolytes, which have been well studied for Zn-based batteries, yielding ZnO as the discharge product, which precipitates on the anode. However, Zn anode [ $E^0 = -0.76$  V vs. standard hydrogen electrode (SHE), and  $< -1.2$  V vs. SHE in strongly alkaline solutions] shows low coulombic efficiency (CE, below 93% and sometimes even  $< 50\%$ ) in alkaline electrolyte due to H<sub>2</sub> evolution reaction (HER) and decay of Zn anode (Zn dendrite and 'dead ZnO'),<sup>[3]</sup> limiting the cycle life of alkaline Zn-O<sub>2</sub> batteries to  $< 100$  cycles.<sup>[4]</sup> In addition, alkaline electro-

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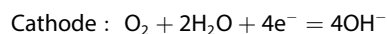
lytes react with  $\text{CO}_2$  from the air to yield  $\text{K}_2\text{CO}_3$  that passivates the cathode. Recent research effort has achieved Zn anode CE over 95% in three types of electrolytes,<sup>[3b]</sup> namely ionic liquid analogues (including water-in-salt electrolytes and deep-eutectic electrolytes),<sup>[3a,5]</sup> neutral or mildly acidic dilute aqueous electrolytes,<sup>[6]</sup> and organic electrolytes.<sup>[7]</sup>

Zn– $\text{O}_2$  batteries with water-in-salt electrolytes {1 m zinc bis(trifluoromethanesulfonyl)imide  $[\text{Zn}(\text{TFSI})_2]$  + 20 m LiTFSI, where m is molality, mol/kg} were reported to yield ZnO as the discharge product.<sup>[3a,5a]</sup>

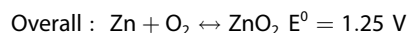
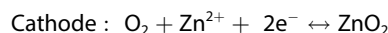
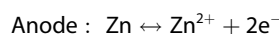


The formation and removal of ZnO on the cathode have been confirmed by X-ray diffraction (XRD), Raman spectra, energy-dispersive X-ray spectroscopy (EDX) and scanning electron microscope (SEM) (Figure 1b, d, e and f).<sup>[3a,5a]</sup> The ZnO crystals exhibit well-defined hexagonal prism morphology (Figure 1, d and g).<sup>[3a]</sup> Up to 200 cycles was demonstrated (Figure 1a).<sup>[3a,5a]</sup>

Recently, a new neutral dilute aqueous electrolyte of 1 m zinc trifluoromethanesulfonate  $[\text{Zn}(\text{OTf})_2]$  in  $\text{H}_2\text{O}$  was proposed as a Zn– $\text{O}_2$  electrolyte that enables  $\text{ZnO}_2$  generation in discharge.<sup>[2]</sup> Conventionally, Zn– $\text{O}_2$  batteries with neutral dilute aqueous electrolyte generates  $\text{OH}^-$  during ORR:



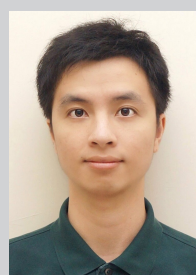
which then reacts with  $\text{ZnSO}_4$  to trigger precipitation of a low-specific-energy discharge product, e.g.  $\text{ZnSO}_4[\text{Zn}(\text{OH})_2]_3 \cdot 0.5\text{H}_2\text{O}$ ,<sup>[8]</sup> resulting in lower energy density. In the new electrolyte, according to multiscale simulations,<sup>[2]</sup> the new chemistry was enabled by the hydrophobicity of the  $\text{OTf}^-$  anions, which results in a water-poor and  $\text{Zn}^{2+}$ -rich inner Helmholtz layer in the cathode, in which a  $\text{Zn}^{2+}$ -assisted  $2\text{e}^-/\text{O}_2$  ORR pathway without water participation is favored over the conventional  $\text{H}_2\text{O}$ -involving  $4\text{e}^-/\text{O}_2$  ORR pathway:



The reversible formation and removal of fiber-shaped  $\text{ZnO}_2$  were comprehensively supported by SEM (Figure 1h), XRD (Figure 1h), EDX, Raman and XPS. Operando pressure measurement (Figure 1i) confirms  $\sim 2 \text{ e}^-/\text{O}_2$  ratios in both ORR and OER. The stable cycling voltage profiles (Figure 1j) indicate high reversibility of this new chemistry. Up to 320 cycles were demonstrated at  $1.0 \text{ mA cm}^{-2}$ .<sup>[2]</sup>

With the primary discharge product identified as  $\text{ZnO}/\text{ZnO}_2$ , more characterization is needed to quantify the yield of  $\text{ZnO}/\text{ZnO}_2$  and probe parasitic reactions that degrade the cathode. For instance, Wang *et al.*, Chen *et al.* and Sun *et al.* observed that the discharge voltage of Zn– $\text{O}_2$  cells gradually decayed upon cycling (Figure 1a and c), which was attributed to catalyst aggregation and loss of active area on the cathode due to passivation by a solid layer of by-products, as supported by SEM.<sup>[3a,5a]</sup> Besides, cathode degradation by side reactions is also possible. In fact, Raman spectrum of the Zn– $\text{O}_2$  cathode cycled in 1 m zinc  $\text{Zn}(\text{TFSI})_2$  + 20 m LiTFSI electrolyte indicated the formation of  $\text{Li}_2\text{CO}_3$  (Figure 1b).<sup>[5a]</sup> This could result from reaction with  $\text{CO}_2$  from the air, or degradation of the  $\text{TFSI}^-$  ion on the cathode triggered by potential intermediates in the electrochemical formation and decomposition of ZnO, e.g. superoxide and singlet  $\text{O}_2$  ( $^1\text{O}_2$ ), which has been reported in Li– $\text{O}_2$  batteries to be associated with electrolyte degradation<sup>[9]</sup> (discussed in detail in Section 3). In the Zn– $\text{O}_2$  cells cycled in  $\text{ZnCl}_2 \cdot 2.33\text{H}_2\text{O}$  electrolyte, the authors reported  $\text{Cl}_2$  gas evolution at a yield of  $< 0.8\%$ .<sup>[3a]</sup> Future studies could explore suppressing  $\text{Cl}_2$  evolution by reducing the charge potential with catalyst, as  $\text{Cl}_2$  evolution not only irreversibly decomposes electrolyte salts but also corrodes cell components including cell case and current collectors.

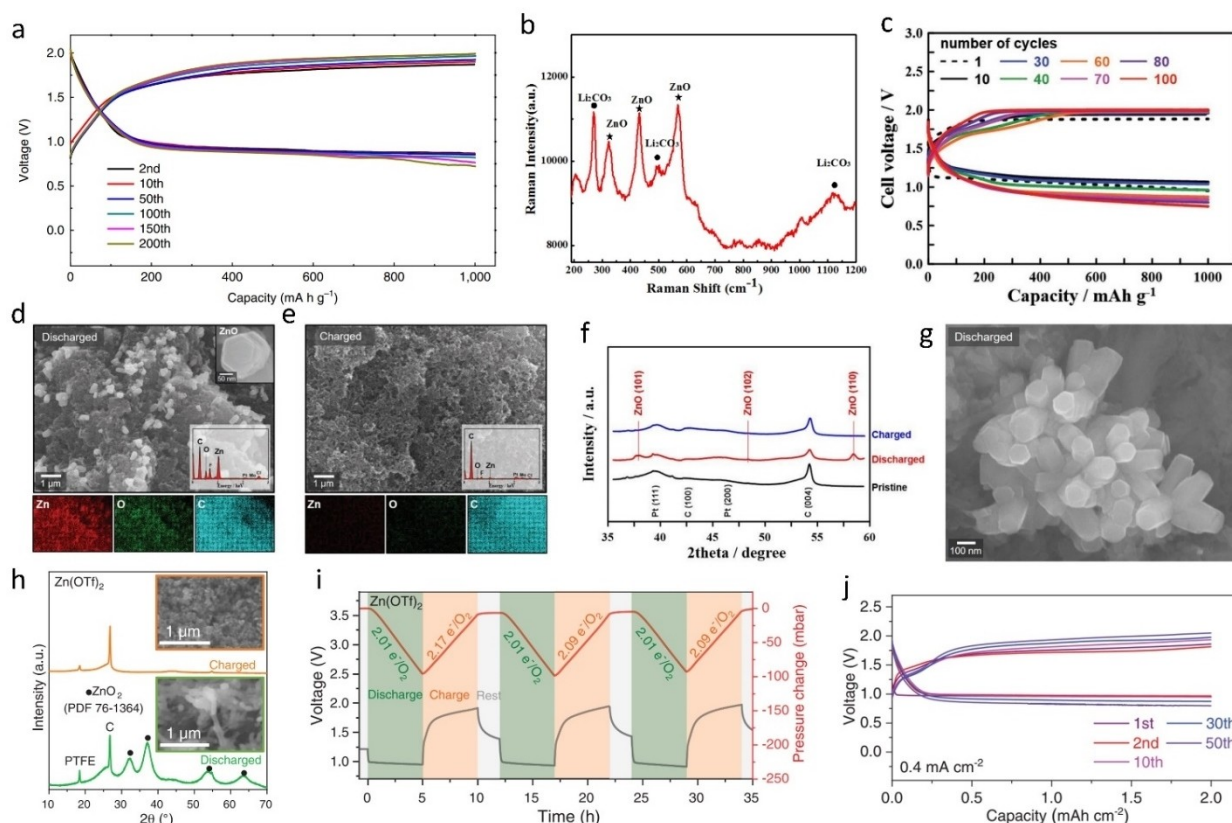
Identifying the discharge intermediate(s) and quantifying their solubility are important for improving the rate capability and discharge capacity of Zn– $\text{O}_2$  batteries. The discharge overpotential was  $\sim 800 \text{ mV}$  at current densities of  $0.035 \sim$



Zhuojian Liang received his Ph.D. at the Chinese University of Hong Kong (CUHK) in 2018. Since joining Prof. Yi-Chun Lu's group at CUHK in 2013, he has been actively developing mechanistic understanding of Li– $\text{O}_2$  batteries using operando techniques and exploring novel solutions to the instabilities. His current research focuses on developing new electrolytes and novel battery configurations for metal– $\text{O}_2$  batteries.



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**Figure 1.** Mechanistic characterization of Zn–O<sub>2</sub> batteries. a) Cycling performance of the Zn/O<sub>2</sub> battery with 1 m Zn(TFSI)<sub>2</sub> + 20 m LiTFSI aqueous electrolyte at a current density of 50 mA/g (areal capacity 0.7 mAh/cm<sup>2</sup>). b) Raman spectrum of the cycled cathode at the discharge state of the Zn/O<sub>2</sub> cell. Reproduced from Ref. [5a] with permission. Copyright 2018, Springer Nature. c) Cycling performance of the Zn-air cell with ZnCl<sub>2</sub> · 2.33H<sub>2</sub>O electrolyte at a current density of 500 mA/g (areal capacity 0.5 mAh/cm<sup>2</sup>). SEM images, EDX analysis, and elemental mapping of the Pt/C cathodes at d) the discharge state and e) the charge state. f) XRD patterns of Pt/C cathodes recovered from the Zn-air cell. g) SEM images of the discharged Pt/C cathode. Reproduced from Ref. [3a] with permission. Copyright 2019, Wiley-VCH. h) XRD patterns and SEM images of air cathodes obtained after discharge and recharge in Zn(OTf)<sub>2</sub> electrolyte. i) Pressure change of Zn–O<sub>2</sub> cells during galvanostatic cycling at 0.4 mA/cm<sup>2</sup>. j) Galvanostatic cycling at a fixed capacity of 2.0 mAh/cm<sup>2</sup> and a current density of 0.4 mA/cm<sup>2</sup>. Reproduced from Ref. [2] with permission. Copyright 2021, American Association for the Advancement of Science.

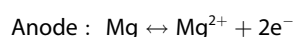
4.0 mA/cm<sup>2</sup> in these initial studies.<sup>[2–3,5a]</sup> This could be limited by multiple factors including the solubility of intermediate(s) and/or diffusivity of O<sub>2</sub>. The solubility of ORR intermediates, which depends on electrolyte properties, governs the competition between the surface-discharge-pathway and the solution-discharge-pathway. In the solution pathway, intermediates dissolve into the electrolyte after their formation on the cathode and diffuse to the surface of existing discharge product for further reactions, which frees the cathode surface from blocking by discharge products, leading to higher discharge current densities and larger discharge capacity. The three-dimensional growth of ZnO crystal indicates some solubility of the intermediate(s), yet the small dimension of a few hundreds of nanometers suggests that the solubility is low. Improving the solubility of intermediate(s) by the use of solvent or additives with higher donor and/or acceptor numbers,<sup>[10]</sup> e.g. crown ethers<sup>[11]</sup> and borates,<sup>[12]</sup> could improve the discharge performance. Under large current densities, the diffusion rate of O<sub>2</sub> from the electrolyte/gas interface to the cathode surface could be limiting and contributes to overpotential. The use of solvents and additives that display high O<sub>2</sub> solubilities and

diffusivities, e.g. ionic liquids<sup>[13]</sup> and perfluorinated carbon liquids,<sup>[14]</sup> are worth exploring.

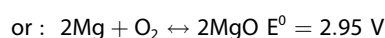
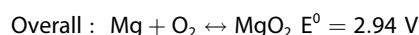
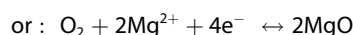
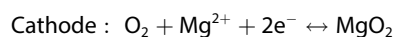
## 2.2. Mg–O<sub>2</sub> Batteries

Aqueous Mg–O<sub>2</sub> batteries are primary batteries<sup>[15]</sup> due to the irreversibility of aqueous Mg anode, which reacts at a substantially lower potential than HER ( $E^0 = -2.37$  V vs. SHE). To enable reversible Mg–O<sub>2</sub> batteries, electrolytes that allow reversible plating and stripping of Mg metal must be used. High Mg anode CE of 98%~100% are reported in multiple types of electrolytes, including simple-salt-based electrolytes,<sup>[16]</sup> Mg–Cl-complex-based electrolytes<sup>[17]</sup> and boron-based electrolytes,<sup>[18]</sup> whose application in rechargeable Mg–O<sub>2</sub> batteries could be explored.

Theoretical studies have shown that both MgO<sub>2</sub> and MgO are possible discharge products at ambient temperature and pressure.<sup>[19]</sup>



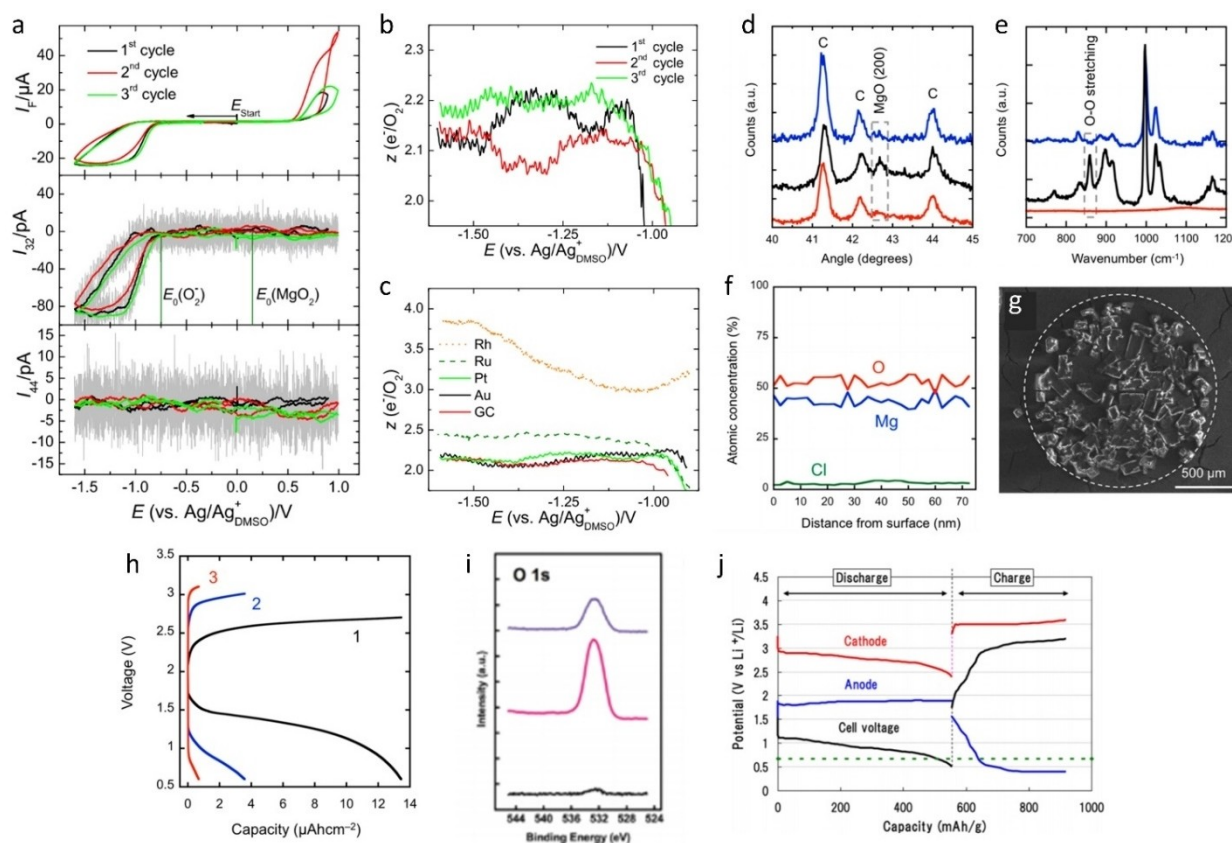




Investigations on ORR mechanism in simple-salt-based electrolyte agree with these theoretical studies. In the electrolyte of  $\text{Mg}(\text{ClO}_4)_2$  in dimethyl sulfoxide (DMSO), Reinsberg *et al.* studied the influence of electrode material on ORR and OER mechanisms using rotating ring-disc electrode (RRDE) and *operando* electrochemical mass spectrometry (OEMS) (Figure 2a–c).<sup>[20]</sup> In the cathodic scan on RRDE (Figure 2c), they found that ORR proceeded at  $2\text{e}^-/\text{O}_2$  on GC, Au and Pt, which agrees with the formation of  $\text{MgO}_2$ , and  $\sim 2.4\text{e}^-/\text{O}_2$  on Ru and up to  $\sim 3.8\text{e}^-/\text{O}_2$  on Rh, which suggest additional concurrent formation of MgO or by-products. They discovered an increase of collection efficiency at the ring with increasing rotation rate, which implies chemical reactions after the initial step of ORR, e.g. further reduction or precipitation of the discharge

intermediate and/or product, or other side reactions.<sup>[20]</sup> In the anodic scan, despite the detection of strong oxidative current, neither  $\text{O}_2$  nor parasitic gases like  $\text{CO}_2$  were observed (Figure 2a), which indicates that the discharge product(s) could not be reversibly oxidized on Pt in DMSO. These findings were further corroborated by Fischer *et al.*,<sup>[21]</sup> who found evidence of  $\text{MgO}_2$  on the discharged Pt and Au cathodes with XPS ( $\text{O}_2^{2-}$  peak in O 1s spectra and an O:Mg ratio of 2.2) at a high yield of 96%, as well as peaks of by-products such as  $\text{MgCl}_2$  and dimethyl sulfone ( $\text{DMSO}_2$ ).

$\text{Mg}-\text{O}_2$  reaction products vary significantly between  $\text{Mg}-\text{Cl}$ -complex-based electrolytes. Vardar *et al.* identified the discharge product as a mixture of  $\text{MgO}_2$  and  $\text{MgO}$  in the all-phenyl-complex (APC) electrolyte [ $\text{PhMgCl} + \text{Al}(\text{OPh})_3$  in tetrahydrofuran (THF)].<sup>[19a]</sup> They observed the formation and decomposition of MgO on the cathode with XRD and Raman (Figure 2d–e). Auger electron spectroscopy (AES) confirmed the presence of Mg and O in the discharge products, amounting to a  $\text{MgO}_2$ :MgO molar ratio of 1:3.3 (Figure 2f). The large particle size ( $\sim 200 \mu\text{m}$ ) observed via SEM (Figure 2g) indicates that either the discharge intermediate is highly soluble or that the discharge product is conductive, both of which are beneficial



**Figure 2.** Mechanistic characterization of  $\text{Mg}-\text{O}_2$  batteries. a) CV and OEMS results on a Pt cathode with  $\text{Mg}(\text{ClO}_4)_2$  - DMSO electrolyte. The number of electrons transferred per oxygen molecule on b) Pt and c) multiple cathodes. Reproduced from Ref. [20] with permission. Copyright 2016, Elsevier. d) XRD pattern of control (red), discharged (black) and recharged (blue) carbon cathodes in APC electrolyte. e) Raman spectra collected from control (red), discharged (black), and recharged (blue) carbon cathodes. f) AES depth profile of the discharge product. g) SEM images of the discharged cathode. h) Discharge-charge cycles for a  $\text{Mg}-\text{O}_2$  cell at  $5 \mu\text{A}/\text{cm}^2$ . Reproduced from Ref. [19a] with permission. Copyright 2015, American Chemical Society. i) O 1s XPS spectra of the as-prepared (black traces), discharged (pink traces) and fully recharged (violet traces) carbon paper cathodes in the electrolyte with dual mediators. Reproduced from Ref. [22] with permission. Copyright 2016, Royal Society of Chemistry. j) Discharge-charge profile of a  $\text{Mg}-\text{O}_2$  cell in a three-electrode cell with a  $\text{Li}/\text{Li}^+$  reference electrode. Reproduced from Ref. [23] with permission. Copyright 2014, American Chemical Society.

for realizing large discharge capacity. Both AES and SEM showed that the discharge product was only partly removed during charge, suggesting the existence of side reaction(s). On the contrary, although the identical porous carbon cathodes were used, no  $\text{MgO}_2$  or  $\text{MgO}$  was found in  $\text{Mg-Al-chloride-complex}$  (MACC) electrolyte [ $\text{MgCl}_2 + \text{AlCl}_3$  in dimethoxyethane (DME)] after discharge.<sup>[24]</sup> Instead, XRD revealed only precipitations of  $\text{MgCl}_2$  and  $\text{Mg}(\text{ClO}_4)_2$  on the discharged cathode. The different discharge products in APC and MACC electrolyte indicate that the discharge reaction is sensitive to not only the type of the working ion (e.g.  $\text{Mg-Cl}$  complex or  $\text{Mg}^{2+}$ ) but also to other existing species in the electrolyte.

The use of redox mediators has been reported in simple-salt-based electrolytes.<sup>[22–23,25]</sup> Mediators are soluble and electrochemically active species that act as charge carriers in the electrolyte during cycling. For example, in metal- $\text{O}_2$  batteries, a charge mediator facilitates the oxidation of metal (su)(per)oxides into  $\text{O}_2$ . During charge, the mediator is firstly oxidized, diffuses to the discharge products, and oxidizes them to evolve  $\text{O}_2$ , during which the mediator is reduced back to its original state. Similarly, a discharge mediator facilitates the reduction of  $\text{O}_2$  to (su)(per)oxides. The observed (dis)charge potential of the cell is determined by the redox potential of this (dis)charge mediator, which should be higher than the reversible potential of OER (or lower than that of ORR) to provide driving force of the chemical redox reactions. Shiga *et al.* first showed that a  $\text{Mg-O}_2$  battery with  $\text{Mg}(\text{ClO}_4)_2$  in DMSO electrolyte can discharge to form  $\text{MgO}$  as revealed by time-of-flight mass spectrometry (ToF-MS), however, the cell could not be recharged.<sup>[25]</sup> They showed that the addition of  $\text{I}_2$  into a  $\text{MgO}$  suspension could dissolve  $\text{MgO}$  to evolve  $\text{O}_2$ , and attributed this to the oxidation of  $\text{MgO}$  by  $\text{DMSO-I}_2$  complex. With  $\text{I}_2$  in the electrolyte, charging voltage was reduced while  $\text{O}_2$  was detected by gas chromatography mass spectrometry (GCMS). As a result, the cell could sustain four galvanostatic cycles. The authors further applied 2,2,6,6-Tetramethyl-1-piperidinyloxy (TEMPO) as a charge mediator with  $\text{Mg}(\text{TFSI})_2$  in diethyl-methyl-methoxymethylpiperidium bis(trifluoromethanesulfonyl)imide (DEMETFSI) electrolyte, in which the discharge products are yet to be identified. To prove the chemical reactivity, they showed that  $\text{TEMPO}^+$ -containing solution could dissolve  $\text{MgO}$  to evolve  $\text{O}_2$ . They further showed that charging with TEMPO lowered the voltage. Consistently, EDX mapping revealed a lower Mg coverage on the charged cathode and  $\text{O}_2$  evolution was detected using GCMS. To further facilitate discharge reaction, Dong *et al.* proposed using dual redox mediators {1,4-benzoquinone (BQ) and 5,10,15,20-tetraphenyl-21H,23H-porphine cobalt(II) [ $\text{Co}(\text{II})\text{TPP}$ ]} for rechargeable  $\text{Mg-O}_2$  batteries.<sup>[22]</sup> Cycling with dual mediators reduced overpotentials for both discharge and charge. Pressure in the cell reduced during discharge and increased during charge, and X-ray photoelectron spectroscopy (XPS) of the discharged cathode revealed  $\text{MgO}_2$  as the discharge product (Figure 2i).<sup>[21–22]</sup> These three studies demonstrated  $\text{MgO}_2/\text{MgO}$  and  $\text{O}_2$  as possible products of discharge and charge, respectively. However, some cycling capacity could be contributed by the mediators alone. If the charge mediator is electrochemically oxidized but fails to

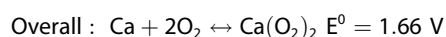
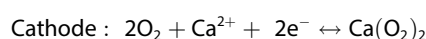
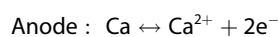
oxidize the discharge product, the charged mediator could be electrochemically reduced in the subsequent discharge (contributing capacity). Based on theoretical standard potentials, neither  $\text{MgO}_2$  (2.94 V vs.  $\text{Mg}/\text{Mg}^{2+}$ ) nor  $\text{MgO}$  (2.95 V vs.  $\text{Mg}/\text{Mg}^{2+}$ ) could be oxidized by  $\text{I}_3^-/\text{I}_2$  (around 2.0 V~2.5 V vs.  $\text{Mg}/\text{Mg}^{2+}$ )<sup>[26]</sup> or  $\text{TEMPO}/\text{TEMPO}^+$  (~2.8 V vs.  $\text{Mg}/\text{Mg}^{2+}$ ).<sup>[27]</sup> Future studies using quantitative techniques would help to understand the mediation mechanisms.

The low discharge voltage of the  $\text{Mg-O}_2$  battery (~2.0 V, Figure 2h) significantly deviates from the reversible potentials of both  $\text{MgO}_2$  (2.94 V vs.  $\text{Mg}/\text{Mg}^{2+}$ ) and  $\text{MgO}$  (2.95 V vs.  $\text{Mg}/\text{Mg}^{2+}$ ).<sup>[19a]</sup> The oxygen electrode potential could be better revealed by using three-electrode cell, which could differentiate the overpotential contribution between anode and cathode.<sup>[23,25]</sup> For instance, Figure 2j shows that the overpotential of the Mg anode could significantly dominate the  $\text{Mg-O}_2$  cells, preventing accurate understanding of the oxygen electrode potential.

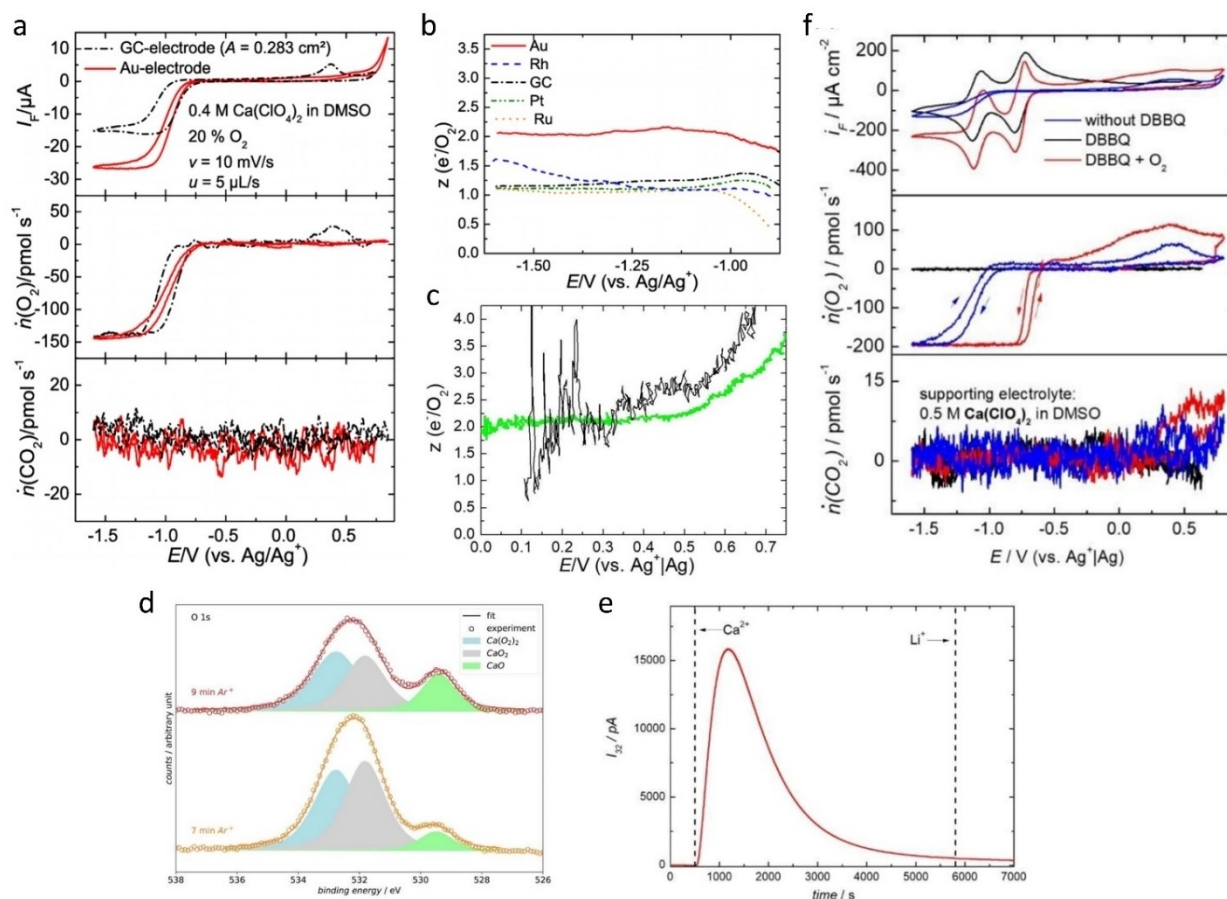
### 2.3. Ca- $\text{O}_2$ Batteries

While reversible plating and stripping of Ca metal ( $E^0 = -2.87$  V vs. SHE) are challenging, recent efforts have realized facile reactions with CE up to 96% in organic electrolytes, enabling the exploration of rechargeable batteries with a Ca metal anode.<sup>[28]</sup>

Various studies have found that ORR in the presence of  $\text{Ca}^{2+}$  yields superoxide or oxide depending on cathode materials. Reinsberg *et al.* studied ORR and OER using RRDE and OEMS.<sup>[29]</sup> On multiple materials, i.e. GC, Pt, Ru and Rh, they found that ORR proceeded at  $\sim 1\text{e}^-/\text{O}_2$  (Figure 3b), and yielded discharge product that could be oxidized back to  $\text{O}_2$  at  $\sim 1\text{e}^-/\text{O}_2$  (Figure 3a), which implies the formation of  $\text{Ca}(\text{O}_2)_2$  with a high reversibility.<sup>[29a]</sup>



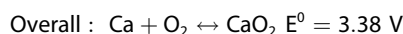
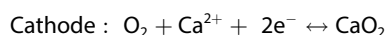
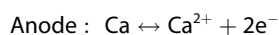
The onset potentials of ORR and OER were  $\sim -0.8$  V and  $\sim 0.2$  V vs.  $\text{Ag}/\text{Ag}^+$  [ $\sim 2.6$  V and  $\sim 3.6$  V vs.  $\text{Ca}/\text{Ca}^{2+}$ , based on  $E(\text{Ag}/\text{Ag}^+) = 3.36$  V vs.  $\text{Ca}/\text{Ca}^{2+}$  they used], respectively (Figure 3a). Interestingly, the ORR onset potential is substantially higher than the reversible potential calculated from thermodynamic data, which could result from the stabilization of  $\text{O}_2^-$  by the cation.<sup>[30]</sup> The high roundtrip overpotentials of around 1 V were attributed to the formation of contact ion-pair between  $\text{Ca}^{2+}$  and  $\text{O}_2^{2-}$ . The discharge product was fairly soluble, with  $\sim 90\%$  dissolved in the electrolyte and only  $\sim 5\%$  deposited on the cathode under the reported testing conditions.<sup>[29]</sup> The remaining 5% of reduced oxygen species was decomposed, which was found to depend on the water content of the electrolyte.<sup>[29]</sup> The ratio of dissolved superoxide could be affected by the experimental conditions including the electro-



**Figure 3.** Mechanistic characterization of Ca-O<sub>2</sub> batteries. a) CV and OEMS results on GC and Au cathodes with Ca(ClO<sub>4</sub>)<sub>2</sub> - DMSO electrolyte. The number of electrons transferred per oxygen molecule b) on multiple cathodes during reduction and c) on Au during oxidation. Reproduced from Ref. [29a] with permission. Copyright 2016, American Chemical Society. d) O 1s XPS spectra collected from the discharged Pt electrode after Ar<sup>+</sup> treatment. e) OEMS signal of the gas evolution from a stirred DMSO solution containing KO<sub>2</sub> after adding Ca(ClO<sub>4</sub>)<sub>2</sub> and LiClO<sub>4</sub>. f) CV and OEMS data on an Au cathode in electrolytes containing DBBQ as a discharge mediator. Reproduced from Ref. [29b] with permission. Copyright 2020, The Authors.

lyte volume, electrode surface area, and actual discharge capacity. With a noticeable solubility of discharge product, the Ca-O<sub>2</sub> system based on Ca(O<sub>2</sub>)<sub>2</sub> could behave like the K-O<sub>2</sub> system based on KO<sub>2</sub>, whose coulombic efficiency grows with cycling owing to the gradual saturation of KO<sub>2</sub> in the electrolyte.<sup>[31]</sup>

The discharge product of the Ca-O<sub>2</sub> battery depends on the use of electrode materials. On the Au cathode, ORR proceeded at ~2e<sup>-</sup>/O<sub>2</sub> from 2.6 V vs. Ca/Ca<sup>2+</sup> (Figure 3, a and b), consistent with the formation of CaO<sub>2</sub>.<sup>[29a]</sup>



The discharge products were also partly soluble.<sup>[29]</sup> The discharge products were oxidized back to O<sub>2</sub> at ~2e<sup>-</sup>/O<sub>2</sub> at low overpotentials but increased to >3e<sup>-</sup>/O<sub>2</sub> at high overpotentials (Figure 3c), probably due to side reactions that occurred during oxidation of the discharge product or by-products. XPS of the

discharged cathode indicates that the discharge products contained O species of various chemical states, including superoxide, peroxide and oxide (Figure 3d), along with S species (attributed to the decomposition of DMSO by Pt or <sup>1</sup>O<sub>2</sub>), and were covered by a carbon-rich passivating layer, which can be attributed to CO<sub>3</sub><sup>2-</sup> salts resulting from side reactions involving superoxide and <sup>1</sup>O<sub>2</sub>.<sup>[9,29b]</sup>

We notice that the ORR mechanism appears to depend on the concentration of Ca<sup>2+</sup>. The above experiments were conducted with 0.1 M ~ 0.4 M Ca(ClO<sub>4</sub>)<sub>2</sub> in DMSO. When increasing the concentration of Ca(ClO<sub>4</sub>)<sub>2</sub> to 1.0 M, mixing of the electrolyte with a KO<sub>2</sub> solution led to immediate O<sub>2</sub> evolution (Figure 3e). This indicates that O<sub>2</sub><sup>-</sup> disproportionation can be catalyzed in the bulk solution by 1.0 M Ca<sup>2+</sup> in addition to specific cathode materials such as Au.

In the above chemical mixing experiment (Figure 3e), OER was observed together with the formation of CO, CO<sub>2</sub>, SO<sub>2</sub>, H<sub>2</sub>O and CH<sub>2</sub>O as detected by mass spectrometer,<sup>[29b]</sup> which indicates that the disproportionation reaction is accompanied by the formation of <sup>1</sup>O<sub>2</sub>. <sup>1</sup>O<sub>2</sub> was shown to originate from superoxide disproportionation and attack electrolyte to evolve parasitic gases.<sup>[9c]</sup> This experiment provides insights into the

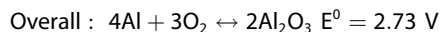
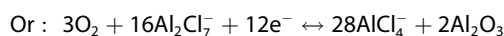
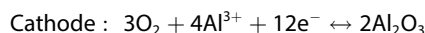
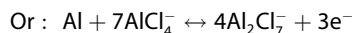
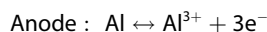


origin of the by-products found in the XPS results (Figure 3d). Future studies including detection of  $^1\text{O}_2$  using fluoroscopy<sup>[9b]</sup> or UV-Vis spectroscopy,<sup>[9d]</sup> are needed to identify the underlying reactions.

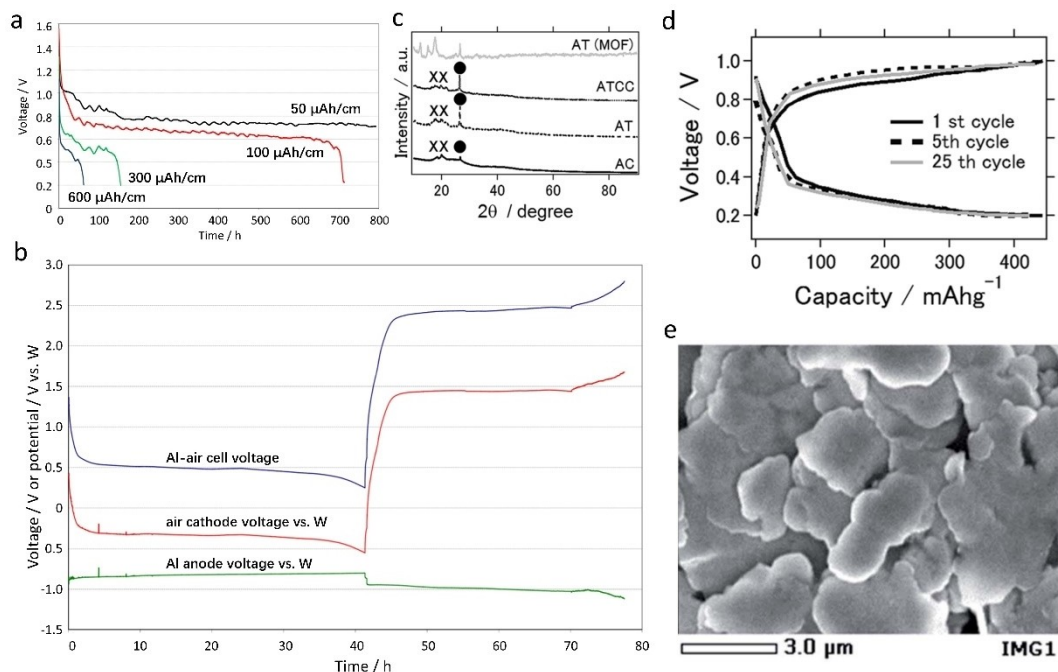
Reinsberg *et al.* reported using 2,5-di-tert-butyl-1,4-benzoquinone (DBBQ) as a discharge mediator for Ca–O<sub>2</sub> batteries (Figure 3f).<sup>[29b]</sup> The DBBQ mediator enhanced both ORR and OER currents. Both ORR and OER onset overpotentials were drastically decreased by  $\sim 0.25$  V, resulting in almost zero overpotential. The authors suggested a mix of  $1\text{e}^-/\text{O}_2$  and  $2\text{e}^-/\text{O}_2$  processes in the DBBQ-mediated ORR based on RDE and OEMS data. Other material characterization techniques including XRD and XPS could be used to verify the final products of the system at various  $\text{Ca}^{2+}$  concentrations.

## 2.4. Al–O<sub>2</sub> Batteries

Aqueous Al–O<sub>2</sub> batteries are primary batteries due to the difficulty in plating Al metal ( $E^0 = -1.66$  V vs. SHE) in aqueous electrolytes, where Al corrosion currents are high in the order of  $10^1$  to  $10^2$  mA/cm<sup>2</sup>.<sup>[32]</sup> Reversible Al plating and stripping with CE as high as 99.7%<sup>[33]</sup> and corrosion rate as low as the  $10^{-3}$  mA/cm<sup>2</sup> scale<sup>[34]</sup> have been demonstrated in Al(TFSI)<sub>3</sub> in acetonitrile,<sup>[35]</sup> ionic liquids<sup>[36]</sup> and ionic liquids analogue.<sup>[33,37]</sup> Plausible reactions of rechargeable Al–O<sub>2</sub> batteries in a simple-salt-based electrolyte or Lewis-acidic ionic liquid that contains chloroaluminate ion ( $\text{Al}_x\text{Cl}_y^-$ ) could be as following:<sup>[34]</sup>



To the best of our knowledge, the first rechargeable Al–O<sub>2</sub> battery was demonstrated in Lewis-acidic ionic liquids [AlCl<sub>3</sub> in 1-ethyl-3-methylimidazolium chloride (EMImCl)].<sup>[34]</sup> Despite a theoretical  $E^0 = 2.73$  V for Al<sub>2</sub>O<sub>3</sub>-based Al–O<sub>2</sub> batteries, the battery displayed low discharge voltages ranging from 0.5 V to 0.75 V at various current densities (Figure 4a). To enable separate measurement of the voltages of anode and cathode, a three-electrode cell was used with a tungsten (W) pseudo-reference electrode, which was calibrated to be 2.45 vs. Al/Al<sup>3+</sup> (0 V vs. W = 2.45 V vs. Al/Al<sup>3+</sup>). The cathode showed an open circuit voltage (OCV), discharge potential, and charge potential at +0.43 V, –0.35 V and +1.4 V vs. W, respectively, translating to 2.88 V, 2.10 V and 3.85 V vs. Al/Al<sup>3+</sup>, respectively (Figure 4b). The OCV agrees well with the theoretical  $E^0 = 2.73$  V for Al<sub>2</sub>O<sub>3</sub>-based Al–O<sub>2</sub> batteries. The low discharge cell voltage largely originates from the Al counter electrode, which showed OCV, oxidation voltage and reduction voltage of –0.93 V, –0.85 V and –1.0 V vs. W, translating to 1.52 V, 1.60 V and 1.45 V vs. Al/Al<sup>3+</sup>. This represents a substantial  $\sim 1.5$  V systematic positive



**Figure 4.** Mechanistic characterization of Al–O<sub>2</sub> batteries. a) Discharge curves of Al-air cells using EMIm/AlCl<sub>3</sub> electrolyte at different current densities. b) Discharge-charge curve at 600 μAh/cm<sup>2</sup> of an Al-air battery equipped with a tungsten pseudo-reference. Reproduced from Ref. [34] with permission. Copyright 2014, Elsevier. c) XRD pattern of the discharged air cathode. Reproduced from Ref. [38] with permission. Published by The Royal Society of Chemistry. d) Discharge-charge curve at 500 μAh/cm<sup>2</sup> of an Al-air battery with a TiC cathode. Reproduced from Ref. [39] under the terms of the Creative Commons License. Copyright 2017, The Authors.

shift in the Al electrode potential. We suggest that this shift could be associated with the formation of a passivation layer on the Al electrode, as revealed in a later work by XPS where the same electrolyte was used.<sup>[38]</sup> The layer could be generated from reaction between fresh Al and O<sub>2</sub>, yielding Al<sub>2</sub>O<sub>3</sub> that could increase the Al anode potential from Al/AlCl<sub>4</sub><sup>−</sup> to Al/Al<sub>2</sub>O<sub>3</sub>. Overall, these observations indicate that the Al–O<sub>2</sub> battery with an ionic liquid electrolyte display a cathode potential that is consistent with the formation of Al<sub>2</sub>O<sub>3</sub>, although accompanied by a high roundtrip cathode overpotential of 1.75 V. In addition, the drastically different potentials between the Al metal electrode in inert and oxidative environment highlights the importance of using three-electrode cells with a non-O<sub>2</sub>-sensitive reference electrode in the study of multivalent metal–O<sub>2</sub> batteries.

Rechargeable Al–O<sub>2</sub> batteries are further studied using the same ionic liquids (EMImCl)<sup>[38–40]</sup> and other ionic-liquid-analogues (AlCl<sub>3</sub> in acetamide and AlCl<sub>3</sub> in urea).<sup>[41]</sup> Stable cycling profiles (Figure 4d) and film-like discharge product morphology (Figure 4e) were reported on a TiC cathode.<sup>[39]</sup> XRD revealed the formation of Al<sub>2</sub>O<sub>3</sub> and Al(OH)<sub>3</sub> phases (Figure 4c) on the cycled carbon cathode,<sup>[38]</sup> which could be the product of ORR. Note that the above-mentioned electrolytes contain chloroaluminate ion (Al<sub>x</sub>Cl<sub>y</sub><sup>−</sup>), which reacts violently with water to yield HCl and white solid.<sup>[42]</sup> Unfortunately, the above works were conducted using air or dry air, which could contain moisture to some extent and introduce Al<sub>2</sub>O<sub>3</sub> or Al(OH)<sub>3</sub> into the discharge product.<sup>[41]</sup> To realize practical use, new electrolyte systems that are not sensitive to air must be developed.

### 3. Experiences Learned from Li–O<sub>2</sub> Batteries

The development of Li–O<sub>2</sub> batteries demonstrates the importance of comprehensive mechanistic investigations. The discharge and charge products should be carefully investigated using quantitative techniques.

In the early stage of research on Li–O<sub>2</sub> batteries, carbonate solvents were widely used due to their successful application in Li-ion batteries. In the discharged cathodes, Li<sub>2</sub>O<sub>2</sub> was found using Raman<sup>[43]</sup> and XRD<sup>[44]</sup> back in 1996. O<sub>2</sub> consumption during discharge was quantified by measuring water displacement to give e<sup>−</sup>/O<sub>2</sub> ratios between 2 and 4, which was attributed to a mixture of Li<sub>2</sub>O<sub>2</sub> and Li<sub>2</sub>O production.<sup>[45]</sup> Gas evolution in charging a Li<sub>2</sub>O<sub>2</sub>-prefilled cathode characterized using OEMS revealed evolution of O<sub>2</sub> and CO<sub>2</sub>, although the amounts were not quantified.<sup>[44]</sup> These findings appear to suggest that the conversion between O<sub>2</sub> and Li<sub>2</sub>O<sub>2</sub> is the main reaction in reported Li–O<sub>2</sub> batteries.

However, more characterizations later found a different conclusion.<sup>[46]</sup> Comprehensive characterization using transmission electron microscopy (TEM), XRD, Fourier-transform infrared spectroscopy (FTIR), nuclear magnetic resonance (NMR), Raman [including surface-enhanced Raman spectroscopy (SERS)] and OEMS revealed that instead of Li<sub>2</sub>O<sub>2</sub>, lithium (alkyl)carbonates are the predominant discharge product in carbonate-based electrolytes.<sup>[47]</sup> The underlying mechanism is that ORR begins

with the formation of the nucleophilic superoxide, which reacts with the carbonate solvent to generate lithium (alkyl) carbonates.<sup>[47d,48]</sup> These by-products are also oxidizable leading to an apparently rechargeable cell, except that their oxidation yields CO<sub>2</sub> and other parasitic gas instead of O<sub>2</sub>.<sup>[47c]</sup>

To realize the intended reduction of O<sub>2</sub> to Li<sub>2</sub>O<sub>2</sub>, solvents that are stable against superoxide, e.g. DME and DMSO, should be used. Li<sub>2</sub>O<sub>2</sub> formation in stable solvents is supported by XRD, Raman, FTIR and in situ XPS.<sup>[47c,49]</sup> More importantly, O<sub>2</sub> and Li<sub>2</sub>O<sub>2</sub> reactions are quantified. As a direct measurement of the degree of stability in discharge-charge cycle, O<sub>2</sub> consumption during discharge and evolution during charge were quantified by the use of pressure gauge and OEMS.<sup>[47c,49b,50]</sup> In complementary, the amount of Li<sub>2</sub>O<sub>2</sub> has also been quantified, using titration and TiOSO<sub>4</sub>-based colorimetry.<sup>[51]</sup> To probe the extent of instability, by-product formation during discharge and charge was characterized by FTIR, NMR and powder XRD (PXRD).<sup>[49b]</sup> A method to quantify by-product by transforming them into CO<sub>2</sub> using Fenton's reagent followed by its quantification using OEMS was proposed.<sup>[52]</sup> Together, these comprehensive investigations picture the reversible transformation between O<sub>2</sub> and Li<sub>2</sub>O<sub>2</sub> via ORR and OER with high certainty.

Alternative to Li<sub>2</sub>O<sub>2</sub>, other stable discharge products have also been reported with advanced characterizations. Using a cathode based on reduced graphene oxide and Ir nanoparticles, LiO<sub>2</sub> is stabilized and precipitates as a crystalline product, as supported by SEM, TEM, high-energy XRD (HE-XRD), Raman, XPS, electron paramagnetic resonance (EPR) and FTIR.<sup>[53]</sup> In a molten salt electrolyte catalyzed by a Ni catalyst, reversible Li<sub>2</sub>O formation and decomposition is characterized using XRD, Raman, SEM, XPS and rotating ring-disc electrode (RDE).<sup>[54]</sup>

Studies on the discharge mechanism are vital for understanding and controlling the discharge stability, rate capability and capacity. During discharge, lithium superoxide (including LiO<sub>2</sub> and the dissociated O<sub>2</sub><sup>−</sup>) were identified as the first intermediate from O<sub>2</sub> reduction with evidence from CV and *in situ* SERS.<sup>[55]</sup> In stable solvents such as ethers, the fate of O<sub>2</sub><sup>−</sup> is strongly influenced by the electrolyte environment.<sup>[56]</sup> It remains as O<sub>2</sub><sup>−</sup> in the presence of only large cations like tetrabutylammonium ion (TBA<sup>+</sup>), and disproportionate into O<sub>2</sub><sup>2−</sup> and O<sub>2</sub> in the presence of small ones like Li<sup>+</sup>, which can be explained using the hard-soft-acid-base (HSAB) theory.<sup>[30,55a]</sup> As an intermediate, LiO<sub>2</sub> displays minor solubility depending on the solvent in the electrolyte, which results in two reaction pathways as probed by RRDE. In the solution-pathway, LiO<sub>2</sub> dissolves into the electrolyte where it disproportionates subsequently. In the surface-pathway, LiO<sub>2</sub> is adsorbed on the cathode where it disproportionates or obtains a second e<sup>−</sup> to yield Li<sub>2</sub>O<sub>2</sub>.<sup>[10a,57]</sup> SEM showed that the competition between these two pathways, which highly depends on the property of solvents, plays an important role in determining the morphology of the discharge product and thus the capacity and rate.<sup>[10a]</sup> This competition is also controllable by the selection of anion, additives and current density.<sup>[10b–d]</sup>

The charging mechanism is also studied in order to address the high charging overpotential. *Operando* XRD character-



ization of  $\text{Li}_2\text{O}_2$  evolution during discharge and charge revealed a two-stage charging feature.<sup>[58]</sup> *In situ* TEM and SEM were applied to study the morphological changes of  $\text{Li}_2\text{O}_2$  and identify active reaction sites, which is important for catalyst design.<sup>[59]</sup> Unlike the discharge reaction in which the intermediate has been well confirmed as  $\text{O}_2^-$ , the charge reaction is less understood and the nature of intermediates remains a topic of debate. A wide variety of techniques including SERS, *operando* XRD, X-ray absorption near edge structure (XANES), *in situ* selected-area electron diffraction (SAED) and RRDE have been applied to probe potential intermediates, e.g.  $\text{O}_2^-$  and  $\text{Li}_2\text{O}_2$ .<sup>[55b,58b,60]</sup> Upon Charging, soluble superoxide has been observed using thin-film RRDE,<sup>[60a,61]</sup> which can disproportionate to form nanocrystalline  $\text{Li}_2\text{O}_2$  and evolve  $\text{O}_2$ . Using aberration-corrected environmental TEM under an oxygen environment,<sup>[60b]</sup>  $\text{LiO}_2$  has been observed as an intermediate phase during part of the charge of a solid-state  $\text{Li}-\text{O}_2$  battery, which implies metastability of  $\text{LiO}_2$  when stabilized by a catalyst such as  $\text{RuO}_2$ <sup>[60b]</sup> or  $\text{Ir}$ .<sup>[53]</sup> More focus should be devoted to revealing the charging mechanism to guide future efforts in reducing the high charging overpotential.

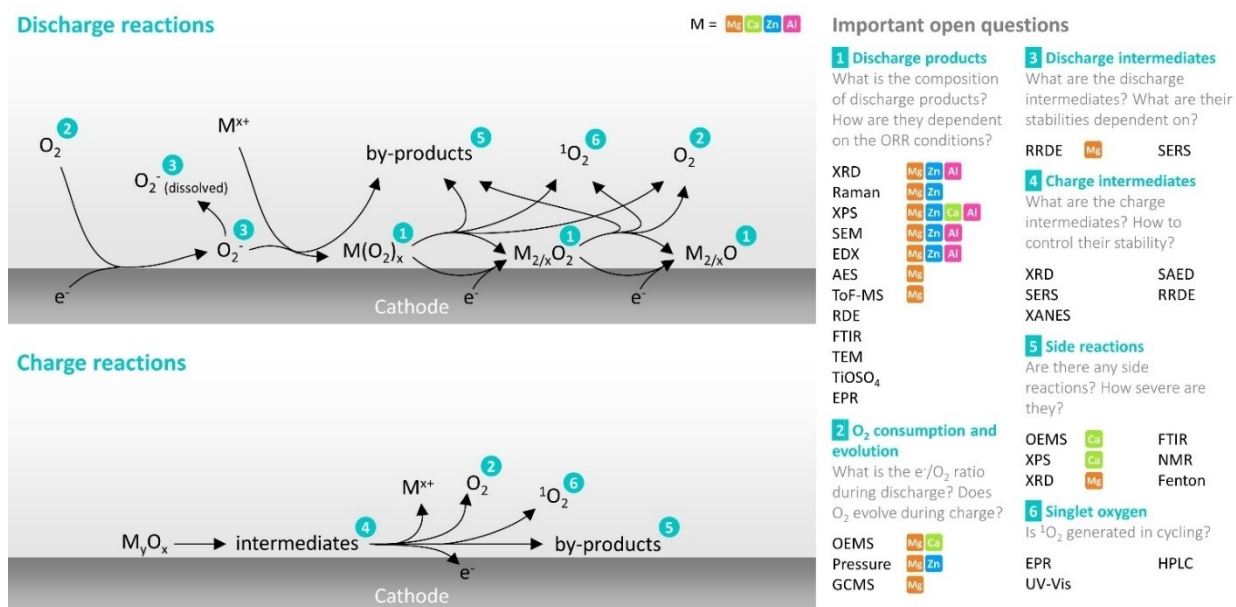
Since  $\text{Li}-\text{O}_2$  batteries display limited cycle life, addressing the instability is one of the most critical steps in advancing this technology to a practical level. A major cause of instability has now been recognized as the formation of  $^1\text{O}_2$ , an excited type of  $\text{O}_2$  gas that displays high reactivity with electrolytes and electrodes.<sup>[9]</sup> It was found to originate from  $\text{O}_2^-$  disproportionation,<sup>[9c]</sup> and oxidation of  $\text{O}_2^-$  and/or  $\text{Li}_2\text{O}_2$ ,<sup>[9a,b]</sup> using multiple quantification techniques including *operando* EPR spectrometry, *operando* UV/Vis spectrometry (fluorescence and absorption) and HPLC, with the help of  $^1\text{O}_2$  traps, e.g. 2,2,6,6-tetramethyl-4-piperidone (4-oxo-TEMP) or 9,10-dimeth-

ylanthracene (DMA).<sup>[9]</sup> Identification of  $^1\text{O}_2$  as a major source of parasitic reaction allows rational development of strategies to address the severe instability to realize reversible  $\text{Li}-\text{O}_2$  batteries including quenchers<sup>[62]</sup> and redox mediators.<sup>[9d,63]</sup>

## 4. Summary and Outlook

To date, ORR and OER in  $\text{Li}-\text{O}_2$  batteries have been intensively characterized by many advanced characterization techniques, and new characterization methods are still continuously emerging. These powerful tools enable researchers to establish a better understanding of the  $\text{Li}-\text{O}_2$  reactions. These understandings serve as theoretical bases and roadmaps in developing new materials for better performances. Compared with the well-established mechanistic studies on  $\text{Li}-\text{O}_2$  batteries, fundamental studies in many rechargeable multivalent metal- $\text{O}_2$  batteries are still in their infancy. The lack of understanding in reaction and degradation mechanisms has hindered further improvement of the performance of these systems. By leveraging the experiences in mechanistic studies on  $\text{Li}-\text{O}_2$  batteries in the past decade, the development of multivalent chemistries can be accelerated. In Figure 5 we have summarized the current status of mechanistic studies on rechargeable metal- $\text{O}_2$  batteries, in the framework of how the mechanistic open questions have been answered, with  $\text{Li}-\text{O}_2$  battery as an example. Future efforts are needed to answer the following crucial open questions:

1. What are the discharge products in these multivalent metal- $\text{O}_2$  batteries? How are they dependent on the ORR conditions, e.g. cathode material, electrolyte composition, current density and  $\text{O}_2$  pressure?



**Figure 5.** A summary of the important open questions and current status of mechanistic studies on rechargeable multivalent metal- $\text{O}_2$  batteries. The techniques that have been used to study mechanistic open questions in  $\text{Li}-\text{O}_2$  batteries are listed. The symbols of multivalent elements next to each technique indicate that this technique has been applied to study the corresponding open question in the  $\text{O}_2$  battery of those elements. Detailed discussions can be found in the corresponding sections.

- Does  $O_2$  evolve during charge? Do other parasitic gases evolve during charging? Are the  $O_2$  consumption and evolution rates consistent with the discharge products?
- What are the discharge intermediates? What is their stability dependent on? Multiple spectroscopic and electrochemical techniques could be applied in synergy to unambiguously identify the discharge intermediates. With these compositions identified, the effect of solvent, salt and additive on the stability and selectivity of these species can be investigated to maximum output capacity and power.
- What are the charge intermediates? How to control their stability to reduce charge overpotential?
- Are there any side reactions? How severe are they? The formation of by-products can be qualitatively and quantitatively analyzed to guide the development of stable electrolytes and electrodes.
- What are the causes of the side reactions?  $O_2$  generation can be probed to help to understand the fate of missing  $O_2$  in some chemistries, e.g.  $Mg-O_2$  and  $Ca-O_2$  batteries.

The development of rechargeable multivalent metal- $O_2$  batteries is critical in achieving batteries with high energy density, low cost, high safety and sustainability. The emergence of new metal-anode-friendly electrolytes has brought great opportunities to further studies on these less-understood chemistries. Guided by the critical insights generated from fundamental mechanistic studies, the full potential of rechargeable multivalent metal- $O_2$  batteries will be realized.

## Acknowledgements

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## Conflict of Interest

The authors declare no conflict of interest.

**Keywords:** metal-oxygen batteries · oxygen electrodes · electrolytes · reaction mechanism · multivalent batteries

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