ecalj manual

<https://github.com/tkotani/ecalj>

**Reference**

Jan2017 Add minimum description for syml generator at Sec.[4.7.1](#sec:getsyml)  
 Aug2019 slight modification for CMD35

# Introduction

The ecalj  is an first-principles electronic-structure calculations package with some unique features. With ecalj , we can do not only standard calculations (LDA/GGA/LDA+U, relaxation of atomic positions), but also the quasiparticle self-consistent GW calculations(QSGW), linear responses (charge and spin), Wannier functions (and U of them).

This is base on an unique one-body problem solver, the PMT method (=the Linearized APW+MTO method) . Thus we identify the QSGW method implemented in ecalj as the PMT-QSGW method. Introduction to the PMT-QSGW is given in Sec.[1.1](#sec:fecalj). Today “QSGW” is accepted as a standard procedure in the electronic structure calculations .

First, see README.org shown at <https://github.com/tkotani/ecalj#ecalj-> (or ecalj/README.org in the package). Free to download ecalj  package from it, and use it. The QSGW code is version controlled by git.

The ecalj  is related to a FP-LMTO package lmv7 seen at ecalj/Document/Manual/CategoryAndToken.org (or html).  
The lmv7 and ecalj are branched at year 2009. After branched, we added new features: simple install and test; all codes are in f90 (no C compiler); new methods, especially PMT-QSGW; MPI parallelization for QSGW; simple usage with automatic setting of default files by python ver2; a small tool to convert VASP POSCASR to ecalj, and so on. PMT-QSGW shows more stable convergence than the previous version, FP-LMTO-GW [[1]](#footnote-23).

 ——————–  
  package mainly consists of two parts. One is (in ecalj/lm7K/), the other is (in ecalj/fpgw/).  
, based on   
We can perform standard calculations such as LDA,GGA,LDA+U, atomic position relaxation, and so on. In addition, the PMT part has an interface to perform GW (and QSGW) calculation: the one-body part can include a given non-local exchange-correlation potential stored in a file sigm.\*. [[2]](#footnote-24). The QSGW calculation is performed by a script gwsc, which has an iteration loop calling the one-body program (lmf) and many-body part (GW part) alternatively. The many-body part generate the file sigm.\*. See Fig.[[gwscpicture]](#gwscpicture) and around.  
, based on   
As the inputs for the calculation, we have to supply the eigenfunctions and the eigenvalues from the one-body part to the GW part. The eigenfuncions re-expanded by the two types of basis functions, the atomic-like argumentation functions in the muffin-tin(MT) spheres, and the plane-waves in the interstitial region, say, the interstitial plane-wave (IPW) hereafter. IPW is defined as the usual plane waves in the interstitial region, but zero within MTs’. The IPWs + ”atomic like functions within MTs” make “a basis set to expand eigenfuncitons”. See Eq.(17) of .

We need another basis set to expand “product of eigenfuncions”. That is, the mixed product bases (MPB)  
, which consists of the two kinds of bases (caution: do not mixed up with the basis set for eigenfuncitons); (i)the local atom-centered functions confined to MT spheres, so-called the product basis; (ii) IPW. The product-basis are calculated from products of solutions to the Schrödinger equation within the MT sphere. The Coulomb matrix , the dynamically screened Coulomb interaction , and so on, are expanded in the MPB. It can virtually span all the space made of product of eigenfunctions (but, in practical calculations, we need to use a small size of the bases to reduce computational time). We include full energy-dependence of . See Sec.3 of .

Recently T.Kotani includes the Wannier function generator, which was originally developed by T.Miyake and H.Kino on top of previous version of GW part. Thus the Wannier functions (including effective interactions) can be generated in the PMT-QSGW. (U parameters in the full-screening and cRPA).

## Uniqueness of the ecalj package.

We will explain two unique points of ecalj .  
**PMT**:  
Central part in an electronic structure packages is one-body problem solver. It means how to calculate eigenvalues/eigenfunctions for a given one-body potential. Inversely, we have to generate new one-body potential for given eigenfunctions/eigenvalues based on the density functional theory (DFT) in the LDA or GGA (In the followings, LDA means both of LDA and GGA). Then we can make the electron density self-consistent by iterations until converged, and obtain total energy of ground states. Then we can calculate atomic forces by perturbation. Based on such an one-body problem solver, we can implement kinds of methods; e.g, dielectric function, magnetic susceptibility, transport and so on. Furthermore, we can implement higher-level approximations such as the QSGW method explained below. An one-body problem solver (in linear methods) are characterized by  
  (i) linear combinations of what basis set to represent eigenfunctions;  
  (ii) how to represent electron density and one-body potential.  
In ecalj , we use the PMT method as the one-body problem solver. The PMT method is a new all-electron full potential method. It uses not only the augmented plane waves (APW) but also the muffin-tin orbitals (MTO) together, in addition to the local orbital (lo’s), to represent the eigenfunctions (no other methods use two kinds of augmented waves together). Thus eigenfunctions are expanded in the linear combinations of the APWs, MTOs, and the lo’s. The formulation is clarified in Ref.. Then the electron density and the one-body potential are given in the “3-components representation”. That is, the electron density (one-body potential) is divided into three components,  
   “smooth part onsite muffin-tin (MT) part counter part”.  
Here the counter part is in order to remove smooth part within MTs. This formalism (Soler-Williams formalism ) is also used in the projected augmented wave (PAW) method such as VASP.

We now usually use highly localized MTOs together with APWs of low energy cutoff ( Ry). [[3]](#footnote-26) I think this is promising not only for efficient DFT/QSGW scheme, but also for kinds of applications in future.

:  
In ecalj , we can perform the GW calculation. The usual GW approximation is so-called “one-shot GW” starting from LDA. It usually only calculates differences between the quasiparticle energies (QPEs) and the LDA eigenvalues by a perturbation (only diagonal part of self energy for the LDA eigenfunctions). Its ability is limited; it may fail when its starting point (eigenfunctions and eigenvalues supplied by LDA) is problematic. This is the reason why we originally develop the QSGW method. The QSGW now becomes popular and taken as a possible candidate to go beyond current limitation of such GW and LDA/GGA . In principle, results given by QSGW do not depend on LDA anymore; the LDA are only used to prepare initial condition for self-consistency iteration cycle of the QSGW calculation [[4]](#footnote-27).

Usually the QPEs obtained by QSGW reproduce experiments better than LDA. For example, the band gap by GGA for GaAs is about 0.5 eV in contrast to the experimental value of 1.69 eV [[5]](#footnote-28). On the other hand, the QSGW predicts about 1.8 1.9eV, a few tenth of eV larger than experiment (for practical use, we sometimes use “hybrid functional between QSGW and LDA” so as to obtain smaller band gap). Even in the case of NiO and so on, the QSGW gives reasonable results (there is a tendency to give a little larger band gaps than experiments). This is in contrast to the case of the one-shot GW applied to NiO, where we can not have good agreement with experiments because the stating points in LDA is problematic.

The ecalj  have other functions. LDA+U, atomic forces and relaxation (in GGA/LDA), core level spectroscopy and so on. In addition, we can calculate dielectric functions and magnetic responses from QPEs and the quasiparticle eigenfunctions given by LDA/QSGW. But total energy in QSGW is still in research (shown total energies in QSGW calculations are dummy now).

The QSGW calculations are very time-consuming; roughly speaking, it takes 10 or more times expensive than usual one-shot GW (although we can reduce computational time by choosing computational conditions). Thus the size of systems which we can treat is limited to ten atom in a cell or something, say, with a node of 16 cores; computation may require a week or so to have reasonable convergence. (heavy atoms require longer computational efforts, light atoms faster; non-magnetic systems are easier. We still have much room to accelerate the method, but not have done yet so much. Minimum MPI parallelization is implemented). The computational effort is in the most time-consuming part of QSGW.

### What do we expect for QSGW?

Let us recall hybrid functionals such as B3LYP, and LDA+U. In hybrid functional methods, we use Vxc = (1-)\*LDA+\*(Fock exchange like term), where is taken to be usually [[6]](#footnote-30). The can be dependent on materials; for metals should be almost zero. For larger band gap insulator, becomes larger. [[7]](#footnote-31) Despite of success of the functional, its ability is limited. For example, it is known that a hybrid functionals fail to describe metals such as bcc Fe. On the other hand, we have LDA+U method which succeeded to describe materials including localized electrons. However, it contains kinds of ambiguity and U is chosen by hand. The important part of the hybrid functional methods and LDA+U is the non-local potential. It is missing in the DFT. As we discussed above, they give some success but not satisfactory. We somehow need to have a method to determine high-quality non-local potential (a substitution of the exchange-correlation potential in LDA). It is the QSGW method.

Note two important aspects of non-local potential (missing character in the local potential used in DFT). One is the onsite non-locality; it is also taken into account by LDA+U model. However, note that relative shift of O(2p) band with respect to the center of 3d band is not in LDA+U. The other is the off-site non-locality (mainly between nearest neighbors), which may relate to LUMO-HOMO gap. A non-local potential can behave a projector which push down only the HOMO states (valence band) to lower energy. This can be in the hybrid functional but not in LDA+U.

In the QSGW, we determine such a non-local potential with the calculation of the GW method, in a self-consistent manner (we repeat GW calculations until converged). We can expect QSGW much more than hybrid methods/LDA+U. Roughly speaking, because the QSGW automatically determine U of LDA+U, or alpha of the hybrid functionals. More accurately speaking, we determine not only but also (the screened Coulomb interaction) self-consistently. Here corresponds to U and alpha. Thus QSGW gives reasonable results even if it is applied to metals such as Fe. For systems with metallic screening, it gives small non-locality (results are close to those of LDA). For systems with large band gap, QSGW gives large enough non-locality (like 0.25\*(Fock exchange)).

Since we now need to treat complex systems, e.g, metal on insulator, it is very essential to treat kinds of materials on a same footing.  
The main purpose of QSGW is to determine an one-particle effective Hamiltonian [[8]](#footnote-32), which describes the quasiparticle picture (or independent-particle picture) for the system we calculate. In other words, QSGW divides the full many-body Hamiltonian into . The screened Coulomb interaction is determined self-consistently in the QSGW iteration cycle.  
In comparison with LDA, we see differences;

* Band gap. QSGW tends to give slightly larger than experiments. It looks systematic.
* Band width. Usually, sp bands are enlarged (except very low density case such as Na). This is the case for homogeneous electron gas. As for localized bands like 3d electrons, they can be narrowed.
* Relative position of bands. e.g. O(2p) v.s. Ni(3d). More localized bands tends to get more deeper. Exchange splitting between up and down (like LDA+U) get larger. In cases such as NiO, magnetic moment become larger; closer to experimental values.
* Hybridization of 3d bands with others. QSGW tends to make eigenfunctions localized.

However, reality is complexed, and not so simple in cases.

## Rule in this manual

* This font is for executable file(program) or shell scripts.
* echo 3hbasfp0 means doing hbasfp0 with the argument ’3’ supplied as the standard input (read(\*,\*) in fortran).
* This font is for files, directories, contents of files, or variables used in codes.
* ctrl.si,rst.si and so on mean the case of Si. You may need to replace the extension si for your case. (this extension is given by user. Lower case, number, and underscore are allowed.) In the followings, ctrl.\* means a file wish such an extension.
* There are files named foobarU and foobarD, which are for up spin (isp=1) and for down spin(isp=2), respectively; for example, SEXU and SEXD. We sometimes use foobarU to denote foobarU and foobarD together.
* **k** vector in the Brillouin zone is called as **q** or **k**.

## What can we do with the ecalj package?

At Feb.2015, what we can do is as follow. We have limited parallelization. (e.g. k point parallel).

* LDA/GGA LDA+U, calculations, atomic forces and relaxation. Spin-orbit is included only for co-linear spin-density cases.
* Quasi-particle(QP) energy in the 1st-iteration from LDA. (one-shot )
* Make band plot for LDA and the QP energies.
* Spectrum function of the self-energy . Life time (imaginary part) of QPs.
* Dielectric function, and its inverse. (including local-field effect or not).
* QP self-consistent (QSGW)
* magnetic susceptibility.
* Wannier function. (not only one-body part, but also effective interaction and cRPA)

# Install

Install and minimum tests are easy; even in a note PC, e.g., we can use gfortran in Ubuntu. In Thinkpad T480s (2018) install may take five minutes, and all tests may take another fifteen minutes.

For productive runs, we may need multi-cores. Current implementation for parallelization by MPI is limited (not so much especially for the dielectric function part yet). Thus, probably, it may be not so efficient to use too many cores.

Follow the instruction of ecalj/README.org.  
or we can see the same one at <https://github.com/tkotani/ecalj/README.org>. We have a command ecalj/InsallAll.ifort and so on. This command installs ecalj and run a series of install tests automatically.

## Binaries and Scripts

Main binaries and Scripts contained in ecalj are

* ctrlgenM1.py  
  Generate default input file ctrl.\*from the structure file ctrls.\*. The latter file only contains information of crystal structure.
* lmfa  
  Spherical atom calculation as initial condition, and core charge.
* lmchk  
  Check atomic positions, crystal symmetry, and computational conditions. Check ID of MuffinTins (MTs of atoms specified by ctrl file is re-ordered by computer.
* lmf and lmf-MPIK LDA/GGA,LDA+U calculations. (or we can use Vxc in QSGW instead). We mainly use lmf-MPI (${\bf k}$-parallel version) instead of lmf.
* PROCAR mode of lmf: Fat band mode.  
  -np 4 lmf-MPIK –mkprocar –band:fn=syml mgo@ gives PROCAR  
  (Try an example  /ecalj/MATERIALS/MgO/. Run ./job at the directory).
* gwsc  
  QSGW calculation
* ,jobUUUUUUtdos,jobUUUUUUpdos@  
  band, fermi surface, tdos, pdos plot.
* Run cu -np 4 10 10 10@. This write down all band energies (around Ef) for 10x10x10 in BZ. Then we can view it with xcrysden as –bxsf fermiup.bxsf@ (usage of xcrysden need to be written here). With –allband added for , we can write all the bands. So, we need this option if you like to write iso-energy surface at high (or low) energy.
* epsPP  
  Dielectric funciton without local field correction (LFC).
* eps  
  Dielectric funciton with LFC
* epsPP  
  Non-interacting transverse spin polarization.
* gw  
  One-shot GW calculation. This also show life-time of QPs (QPU). (we need make it parallelized...)
* genMLWF  
  Wannier functions and matrix elements of on it. A implementation of cRPA included.
* dqpu  
  A small python script to compare QPU.\* files (eigenvalues are compared) numerically.

## tests

Install.ifort run tests at /TestInstall@.  
In the following, :gwUUUUUUlmfh@ means ’siUUUUUUgwUUUUUUlmfh@’ at /TestInstall/@; this test is performed with the Makefile at the directory.

si:gwUUUUUUlmfh/ Results: QPU   
si:gwsc/ : QPU,log.si  
gas:gwsc/ : QPU,log.gas  
nio:gwsc/ : QPU,log.nio  
feUUUUUUepsPPUUUUUUlmfhUUUUUUchipm: : ChiPM\*   
gas:epsUUUUUUlmfh/ : EPS\*  
gas:epsPPUUUUUUlmfh/ : EPS\*

(These are just samples; not for practical calculations)

## Directory structure

ecalj  
├── InstallAll.ifort, Install.gfortran ! install and test  
├── Document/  
├── fpgw/ ! full potential GW code  
├── fpgw/Wannier ! MaxLoc Wannier and cRPA code.  
├── lm7K/ ! PMT method   
├── MATERIALS/ !jobUUUUUUmaterials.py contains examples.   
├── StructureTool/ ! POSCAR converter, and structure tool  
├── GetSyml/ ! Brillouin zone and symmetry lines generator and visualizer.  
├── TestInstall/ ! Install test; this is invoked from Install.\*  
└── TOOLS/ !Tools for developers

# Theory (note)

Except the technical details, we need to know minimum for these theories.

* DFT in LDA/GGA.
* GW
* QSGW

There are literatures as for GW. A recent one is Ref.. In addition, it is better to know the basics of the PMT method (LAPW+LMTO method) . Here is a small note for .  
  
The Green’s function $G({\bf r},{\bf r}',\omega)$ is the central quantity in the GW calculation. In the one-particle theory (mean-field theory/non-interacting case), the Schrödinger eq. is

$$\begin{aligned}
(i \frac{\partial }{\partial t} - HUUUUUU0) \psi({\bf r},t)=0 \label{sch0}\end{aligned}$$

Here is the one-particle Hamiltonian which contains electrostatic potential plus exchange-correlation potential $VUUUUUU{\rm xc}$ (here we don’t care how it is given. This should be static but can be non-local as Fock term). When we have (unkown) source term $J({\bf r},t)$ instead of 0 in the right hand side of , we can calcuate $\psi({\bf r},t)$ by multipling inversion of the operator . This is the (non-interacting) Green funciton $G({\bf r},{\bf r}',\omega)$ defined as

$$\begin{aligned}
(i \frac{\partial }{\partial t} - HUUUUUU0) G
= \delta(t-t')\delta({\bf r}-{\bf r}'). \label{g0def}\end{aligned}$$

As this shows, is the inverse matrix of the operator , (we pay attention to the boundary condition, especially for time-direction: retarted, advanced or time-ordered). Thus we can write . In other words, the Green function is the integration kernel in order to convert a differential equation to an integral equation. This is the same as the case of Poisson equation (Laplacian) for the electrostatic problem;then $1/|{\bf r}-{\bf r}'|$ is the Green function for the conversion (then we calculate electro static field $\phi({\bf r})$ for give source term $\rho({\bf r})$.

In the case of electro static problem, we have two ways to solve the problem. One is the direct method to solve differential equation, the other is using such integration kernel. They are equivalent, but the latter is easy to handle and suitable for numerical calculations (under the assumption of “superposition low”, that is, linear response). Although $G({\bf r},{\bf r}', t-t')$ contains time variables (to describe wave propagation), essentials are the same.  
  
$G({\bf r},{\bf r}',\omega)$ is the quantity which is well defined even in the many-body perturbation theory. It is defined as the expectation value as $\langle 0|V{\psi}({\bf r}t) V{\psi}^\dagger({\bf r}' t')| 0 \rangle$ (this is not accurate. See literatures..). Here we use the second-quantized field-operator . This definition of is a natural extension of the one-particle case.

However, there are differences in the of one-particle case and many-body case. In the one-particle case, we have no interaction between electrons. Thus an electron added to the ground state (which is by filling electrons up the Fermi energy) is the eigenstate. An electron going on can not be an eigenstate, because of the correlation effect and exchange effect.

Let me explain the correlation effect. In contrast to this, one-particle state (=the quasiparticle(QP)) in the many-body theory can be not the eigenstate; it moves in the sea of ohter electrons and holes, which can be excited to other states, or an moving electron can move with polalized cloud of other electrons and holes. Its motion is affected by oher electrons. It will lose energy gradually; the QP can have life time. This may be identified as the correlation effect.

Let me explain the exchange effect. This is due to the Fermi statistics. The electron, which you focus on, can not be distinguished from other electrons. (Thus mu-on can not feel this effect. for mu-on in solids only contains pure correlation effect). From a point of view, this exchange effect is interpreted as a hopping effect, sudden jump from an electron to another elecron (actually the Hartree-Fock theory gives zero effective mass at the Fermi energy for metallic systems).

We can include both of the effects in the GW calculation at the lowest order. Exactly speaking, these two effects are really mixed up. However, in GW (since GW is at the lowest order), these two are clearly separated.

The rigorous equation for is

$$\begin{aligned}
\left(i \frac{\partial }{\partial t} - HUUUUUU0 +
(\Sigma({\bf r},{\bf r}',t-t')-VUUUUUU{\rm xc})\right) G
= \delta(t-t')\delta({\bf r}-{\bf r}'). \label{gdef}\end{aligned}$$

Only the difference is adding $\Sigma({\bf r},{\bf r}',t-t')-VUUUUUU{\rm xc}$. In other words, all unknown effects are pushed into this term. Many body effect are pushed into (downfolded or projected into) the self-energy (=dynamical one-particle potential) $\Sigma({\bf r},{\bf r}',t-t')$. We don’t treat many-body quantities such as $X(\bfrUUUUUU1 tUUUUUU1, \bfrUUUUUU2 tUUUUUU2, \bfrUUUUUU3 tUUUUUU3)$ directly in the calculations.

The method is how to give the $\Sigma({\bf r},{\bf r}',t-t')$ in the lowest order. Because of the long-range property of the Coulomb interaction, we need to have a special technique (not a simple perturbation). $\Sigma({\bf r},{\bf r}',t-t')$ is calculated as the product of , where is (usually) just the non-interacting Green function given in . is the dynamically screened Coulomb interaction calculated in RPA.  
  
Note that completely cancels the effect of $VUUUUUU{\rm xc}({\bf r},{\bf r}')$ since it is included in . Thus this division is not meaningful if we can solve a problem completely. However, what we can do is only the perturbation. Quantities for is taken as the basic quantities. On it, we apply the perturbation. Equivalently we can have a division , where is the many-body Hamiltonian.

In priniple, we can start from any $VUUUUUU{\rm xc}$. However, as long as we use perturbation theory, we have to minimize the effect of $\Sigma({\bf r},{\bf r}',t-t')-VUUUUUU{\rm xc}$. The degree of freedom of the choice of $VUUUUUU{\rm xc}$ is used to minimize it. This is the self-consistent pertubation theory idea for QSGW.  
If $\Sigma({\bf r},{\bf r}',t-t')-VUUUUUU{\rm xc}$ give small effect, we can use the concept of the QP (independent particle picture) as the basis to evaluate physical quantities.

We have to determine best (or optimum) $VUUUUUU{\rm xc}$. How to do it? An idea is that we should determine it so that the size of $\Sigma({\bf r},{\bf r}',t-t')-VUUUUUU{\rm xc}$ should be smallest (how to measure the size?). Or, based on the Landau-Silin’s theory, the QPs contained in should be reproduced by . In anyway, this ends up with the self-consitent petrurbation theory. For trial $VUUUUUU{\rm xc}$ (and electron density), we have ; from we can calculate $\Sigma({\bf r},{\bf r}',t-t')$. Then we have to determine new $VUUUUUU{\rm xc}$ in a method. This is repeated until converged. No unique choice for the method. Thus we testes some possible ways and now we usually use a standard procedure as described in Ref. .

# LDA/GGA calculations and Plots

Calculations are performed by following steps. These steps are detailed in the following sub-sections.

To identify a set of files used for a material we calculate, we use an extension to files. For example, files explained below are with extensions (only lower case allowed) of materials. For example, ctrls.cu and ctrl.cu. In this case cu is the extension. Any extension works. Other possible examples are ctrls.lagao3, ctrl.wgantest1, and so on.

1. Write crystal structure file ctrls.\*, which contains crystal structure. It can be by hand, or convert it from POSCAR (in VASP). There is a tool to convert between POSCAR and ctrls.\* (See ecalj/StructureTool/README.txt for the tool. we have vasp2ctrl and ctrl2vasp. Type them without arguments to see help.)
2. Generate ctrl.\*from ctrls.\*by a script ctrlgenM1.py, [[9]](#footnote-42)

* Here ctrl.\* is the main control file which contains all required information to perform calculations. ctrl.\* contains not only the content of ctrls.\*, but also other information needed for calculations. If necessary, we edit the generated ctrl.\* file before next step.

1. Check crystal structure. lmchkis to confirm the crystal structure (space-group symmetry and so on). lmchkis applied not to ctrls.\*but to ctrl.\*. ctrls.\*never used in the following steps.

* : If a crystal structure is only slightly different from a structure with higher symmetry, the lmchkmay give a wrong crystal symmetry. In such a case, you have to “standardize structure” by VESTA or some other tools. This occurs e.g., when we use a structure numerically relaxed by VASP.

1. Run lmfa (calculations of spherical atoms (MT sites) in the cell). It also calculates core eigenfunctions and valence electron charge to set up initial condition. Then we run main calculation of LDA by lmf. It repeats iterations, and end up with converged results in LDA. Main result (electron density satisfying self-consistency) is stored in restart file rst.\* (binary file). It finished within a second.
2. Run LDA/GGA calculations. We can run the LDA/GGA calculation by lmf or lmf-MPIK (-MPIK means kpoint parallel version).
3. Post processing.  
   Plot energy band, DOS, PDOS, by running scripts. We can use scripts job for band plot (need syml.si file (symmetry line for band plot)). We also have job, job, job and so on for DOS, PDOS, fermi surfaces. Since we use gnuplot to plot them, meanings of obtained data is apparently clear.

## Write crystal structure file, ctrls

Let me show some samples of crystal structure files ctrls.\*.

* ~/ecalj/lm7K/TESTsamples/Cu/ctrls.cu  
  ------from here ------------------  
  % const da=0 alat=6.798  
  STRUC ALAT={alat} DALAT={da}  
   PLAT= 0.0 0.5 0.5 0.5 0.0 0.5 0.5 0.5 0.0  
  SITE ATOM=Cu POS=0 0 0  
  ------to here ------------------
* ecalj/lm7K/TESTsamples/GaAs/ctrl.gaas  
  ------from here ------------------  
  #id = GaAs  
  %const bohr=0.529177 a=5.65325/bohr   
  STRUC  
   ALAT={a}   
   PLAT=0 0.5 0.5 0.5 0 0.5 0.5 0.5 0   
  SITE  
   ATOM=Ga POS=0.0 0.0 0.0  
   ATOM=As POS=0.25 0.25 0.25  
  ------to here ------------------
* ecalj/lm7K/TESTsamples/SrTiO3/ctrls.srtio3   
  ------from here ------------------  
  %const da=0 au=0.529177  
  %const d0=1.95/au a0=2\*d0 v=a0^3 a1=v^(1/3)  
  HEADER SrTiO3 cubic   
  STRUC ALAT={a1} DALAT={da}   
   PLAT=1 0 0 0 1 0 0 0 1  
  SITE  
   ATOM=Sr POS=1/2 1/2 1/2  
   ATOM=Ti POS= 0 0 0  
   ATOM=O POS=1/2 0 0  
   ATOM=O POS= 0 1/2 0  
   ATOM=O POS= 0 0 1/2  
  ------to here ------------------

Lines starting from ’#’ are neglected as comment lines. Lines starting from ’% const’ define variables and set values (in these cases, da, alat, and bohr, and so on). Then the variable alat is referred to as {alat}; in the cu case, {alat} means 6.798. Lines not start from "#" nor "%" are main content in ctrls.\*. [[10]](#footnote-44)

Note that we have two tags of “categories” "STRUC" and "SITE". (“HEADER” tag is also; but it is just for user’s memo shown in console output). These tags should start from the first column. Thus ctrls.\*is divided into multiple “categories”. In a category, we have “tokens” such as ALAT, DLAT, PLAT. These under STRUC category. ALAT+DALAT specify unit of length in this ctrl file. These are in a.u. (= bohr radius=0.529177Å).

The unit cell is given by PLAT (as noted, ALAT+DALAT as unit). In the above example of GaAs, three primitive cell vectors specified by nine numbers after PLAT=; they give three primitive vectors; PLAT1=(0,0,0.5), PLAT2=(0.5, 0.0, 0.5), and PLAT3=(0.5, 0.5, 0). DALAT is convenient to change lattice constant; but it is fixed to be zero here; thus no effect in this example.

Note that SITE category can have multiple ATOM tokens. The number of ATOM token under SITE should be the same as number of atoms in the primitive cell. In the case of GaAs; SITE contain multiple ATOM tokens. POS= just next to ATOM is taken as subtokens under ATOM token. [[11]](#footnote-45) In cases, we specify such subtokens as SITE.

In the SITE category, we place atoms (MT names) in the primitive cell. In these cases we use defaults atomic symbol (MT names) for ATOM. POS is in the Cartesian coordinate (in the unit of ALAT+DALAT).

To test ecalj, you may make a test directory and copy a ctrls.\*to your directory. If you have VESTA and ecalj/StructureTool/ installed, you can see its structure by

$ viewvesta ctrls.cu

(here $ means command prompt).

NOTE: As written in ecalj/README, you have to install VESTA and viewvesta. Then set VESTA= at the top of ecalj/Structure/viewvesta, and make softlink to it. The command viewvesta(~/ecalj/StructureTool/viewvesta.py) generate POSCARUUUUUUcu.vasp first, then send it to VESTA. viewvesta also accept POSCARUUUUUUcu.vasp directly. Except names starting from ctrl and ctrls, viewvesta sends the name to VESTA directly. We need extension ’.vasp’ to recognize it is written in VASP format. We have samples in ~/ecalj/StructureTool/sample.  
A tool vasp2ctrl converts POSCAR.vasp to ctrls.. “–help” show a small help.  
 ecalj/StructureTool/ is not tested well. Not believe it so much... We will fix it on your request. Another possible way is using cif2cell.

If you have a cif file, run

cif2cell foobar.cif -p vasp --vasp-cartesian --vasp-format=5

And convert POSCAR to ctrls. cif2cell is available from github.

In ctrls.srtio3, we use an expression 1/2 to give POS. We can use mathematical expression instead of values. Mathematical expressions such as “ sqrt(...)” are recognized. (instead of 3\*\*2, use 3^2. Use parenthesis, and no space for an expression). We can use default atomic symbols (to check default atom name (MT name) type ctrlgenM1.py --showatomlist). Instead of such default symbols, we can use your own symbol as

SITE  
 ATOM=M1 POS=1/2 1/2 1/2  
 ATOM=M2 POS= 0 0 0  
 ATOM=O POS=1/2 0 0  
 ATOM=O POS= 0 1/2 0  
 ATOM=O POS= 0 0 1/2  
 SPEC  
 ATOM=M1 Z=38  
 ATOM=M2 Z=22  
 ATOM=O Z=8

. Then we have to add extra category SPEC where we set Z number. (You can use Z=37.5 for virtual crystal approximation, however, you can not do it in ctrls now. Edit it in ctrl file. Such a procedure will be explained in other place.xxx)  
This is an example for Antiferro NiO:

#id = NiO  
%const bohr=0.529177 a=7.88  
STRUC ALAT={a} PLAT= 0.5 0.5 1.0 0.5 1.0 0.5 1.0 0.5 0.5  
SITE ATOM=Niup POS= .0 .0 .0  
 ATOM=Nidn POS= 1.0 1.0 1.0  
 ATOM=O POS= .5 .5 .5  
 ATOM=O POS= 1.5 1.5 1.5  
SPEC  
 ATOM=Niup Z=28 MMOM=0 0 1.2 0  
 ATOM=Nidn Z=28 MMOM=0 0 -1.2 0  
 ATOM=O Z=8 MMOM=0 0 0 0

In this case, we define Niup and Nidn sites. These are recognized as Ni atom because of given Z number in SPEC. The subtoken MMOM=Ms,Mp,Md,Mf... re to specify number of magnetic moments ($\muUUUUUUB$) for s,p,d,f channels (difference of up - down electrons within MT sites) as initial condition. In this case, we set n(up)-n(down)=1.2 for Niup site for d channel. Even just one ATOM name is given by yourself, all ATOM in SPEC should be given (in this case SPEC for O should be given).

We can see other samples in ~/ecalj/lm7K/TESTsamples/\*/ctrls.\*. (we also have a sample generator. See later section.) Note that ctrls file is jut in order to generate default ctrl file in the followings. Not from ctrls but from ctrl, we can start calculations. (thus ctrls is not needed if we prepare ctrl file directory).

It is possible to add RELAX= 0 0 1 after SITEUUUUUUATOMUUUUUUPOS; this means structure relaxation along z-axis (also need to set DYN category given in Document/Manual/CategoryAndToken.org). but its defaults are given (but commented out) automatically in the ctrl file generated by the procedure described in the following section). We detail it in other place xxx.  
After ctrl.\* is generated as shown below, we can run a command lmchk to check weather crystal structure is correctly given or not. It finish in a second. It show symmetry information, and so on used in the calculation.

: Positions of atoms are not necessarily fixed by ctrl.\* when you restart calculation with rst.\* file, because atomic positions are read from rst.\*. We need to pay attention when we use DYN option because lmf run may save relaxed atomic positions into the rst.\*. As “lmf –help” shows, si –rs=1,1,1,0,0@ can read atomic position from ctrl file.

## Generate default ctrl from ctrls by ctrlgenM1.py

To run programs of lm7K (lmfa, lmf, lmchk) in ecalj, we need an input file ctrl.\*, which contains not only structures but also other settings. To generate ctrl.\* from ctrls.\*, we have a command "ctrlgenM1.py" (written in python 2.x and call fortran programs(lmfa,lmchk) internally). Two steps required to complete ctrl file: (i) we give reasonable options when we run ctrlgenM1.py. Then (ii) we may edit the ctrl file afterward. In anyway, ctrl file is the starting point of calculations; ctrls is required just in order to generate ctrl.

At first, try ctrlgenM1.py without arguments. It shows help. To generate ctrl from ctrl, type

$ ctrlgenM1.py cu --nk1=8

Here cu specify ctrls.cu. The option –nk1=8 means the number of division of the Brillouin zone for integration. It means 8x8x8 division. If we like to use 8x8x4, we have to supply three arguments –nk1=8 –nk2=8 –nk3=4. The above command gives following console output.

$ ctrlgenM1.py cu --nk1=8  
 === INPUT arguments (--help gives default values) ===   
 --help Not exist  
 --showatomlist Not exist  
 --nspin=1  
 --nk=8  
 --xcfun=vwn !(bh,vwn,pbe)   
 --systype=bulk !(bulk,molecule)  
 --insulator Not exist !(do not set for --systype=molecule)  
  
 ...  
  
 OK! A template of ctrl file, ctrlgen2.ctrl.cu, is generated.

As we see above, options which you specified are shown at the beginning of the console output (in this case –nk1=8). Others such as –nspin=1 are default settings. If we like to perform spin-polarized calculations, we add other option ’–nspin=2’ as

ctrlgenM1.py nio --nspin=2 --nk1=6

(NOTE: In the spin-polarized case, we need to set initial condition of size of magnetic moment at each atoms. Set it in ctrls.\* as in the previous section, or edit MMOM of ctrl file (MMOM=s p d f ...) to be like MMOM=0 0 1.2.). The ctrlgenM1.py generates ctrl file named as ctrlgenM1.ctrl.cu. To do calculations, copy it to ctrl.cu so that lmf can recognize it.

cp ctrlgenM1.ctrl.cu ctrl.cu

## crystal structure checker: lmchk

Do lmchk to confirm that we can let lmf know correct crystal structure. It also show crystal structure informaiton, equivalent sites, site index and so on.

lmchk --pr60 cu (--pr# gives more informations if # is number)

Then it reads ctrl.cu. --pr60 is an option of verbose. Bigger number gives more information.

* Lattice info, Space group symmetry operations (in lmf format), and their generators (these operations can be generated from a few of them.)  
  See SYMGRP in Document/Manual/CategoryAndToken.org. about how to represent the operations.
* Show atomic positions in ctrl file.
* Tabulate MT radius and distance between atomic sites.

(lmchk –help shows help, but difficult to see. Not need to read it first.)  
lmchk is also shows atom (MT site) id (position and class(equivalent positions). This is needed to interpret PDOS.  
:

## ctrl file

It is not necessary to look into ctrl file first, although some details are explained in the generated ctrl file. Please compare obtained results by lmf with those by other packages or literatures; let me know if you find something strange or your questions.

It is necessary to edit ctrl file to use full ability of lmf. For example, LDA+U, atomic position relaxation, core level spectroscopy, Change setting of default MTO and lo, better mixing procedure for stable convergence; higher accuracy, and so on.

But a few of ctrl file is easy to modify. Search these words and read explanations embedded in ctrl file.  
(1)XCFUN  
(choice of XC—it is not need to repeat ctrlgenM1.py). It is also possible to change number of k points for sampling, to modify crystal structure slightly, and so on; all things needed are in ctrl. It is not needed to repeat ctrlgenM1.py again.  
(2)SO  
To obtain correct dispersion around top of valence at point for GaAs, we need to set SO=1 and NSPIN=2. QSGW calculation (by gwsc) do not allow this option now; Thus we run lmf (or lmf-MPIK) with such settings changed in ctrl file, after QSGW is converged.  
lmf --input shows what can we write in ctrl file. But more than half are not for users, but for developers (or irrelevant now).

## Run LDA/GGA calculations, and get convergence

Here we show how to get converged results from a ctrl file.

At first, we need initial guess of charge density. It can be given by a super position of atomic charge density. To obtain the charge density, we solve atoms first. It is by

$ lmfa gaas | tee llmfa

It takes just a few seconds. Here tee is a command of Linux. It keeps console output (standard output) to a file (llmfa in this case).

Then try

$ grep conf llmfa

. Then you see a key point that

conf:SPECUUUUUUATOM= Ga : --- Table for atomic configuration ---  
conf: isp l int(P) int(P)z Qval Qcore CoreConf  
conf: 1 0 4 0 2.000 6.000 => 1,2,3,  
conf: 1 1 4 0 1.000 12.000 => 2,3,  
conf: 1 2 4 3 10.000 0.000 =>   
conf: 1 3 4 0 0.000 0.000 =>   
conf: 1 4 5 0 0.000 0.000 =>   
conf:-----------------------------------------------------  
conf:SPECUUUUUUATOM= As : --- Table for atomic configuration ---  
conf: isp l int(P) int(P)z Qval Qcore CoreConf  
conf: 1 0 4 0 2.000 6.000 => 1,2,3,  
conf: 1 1 4 0 3.000 12.000 => 2,3,  
conf: 1 2 4 3 10.000 0.000 =>   
conf: 1 3 4 0 0.000 0.000 =>   
conf: 1 4 5 0 0.000 0.000 =>   
conf:-----------------------------------------------------

This is an initial electron distribution, and how we divide core and valence. In this case core charge Qcore are “6 electron for s channel=1s,2s,3s and 12 electron for 2s and 3p”. Qcore is treated by frozen core approximation. See Sec.2.5 in Ref.[[kotani2015pmt]](#kotani2015pmt). Qval means electrons for valence s,p,d channels. The valence channels are 4s,4p,4d,4f (if we set EH=s,p,d,f) in this case The int(P)z column is for local orbital. Thus we have 3d treated as local orbital. (ecalj allow add one local orbital per .)

The isp index means spin (1 or 2), since –nspin=1 (when we invoke ctrlgenM1.py) for GaAs, no isp=2 exist. In summary we have 4s,4p,4d,3d,4f as valence. This means we use corresponding number of MTOs and local orbitals.

After lmfa, let us start main calculation.

$ lmf cu

In unix, we ca save console output to llmf by $ lmf cu | tee llmf. As it starts iteration calculations, it shows similar output again and again (it is a little too noisy now). Then you end up with self-consistent result as

......  
 it 8 of 30 ehf= -3304.895853 ehk= -3304.895853  
 From last iter ehf= -3304.895856 ehk= -3304.895855  
 diffe(q)= 0.000003 (0.000007) tol= 0.000010 (0.000010) more=F  
c ehf=-3304.8958531 ehk=-3304.8958529  
 Exit 0 LMF   
 CPU time: 7.024s Mon Aug 19 02:03:19 2013 on

it 8 of 30 means it stop at 8th iteration, although we set maximum number of iteration 30. Note that this number is given by ITERUUUUUUNIT=30 in ctrl.cu). ehf and ehk are the ground state energy in Ry. They are calculated in a little different procedure. Although they are different during iterations, it finally get to be the almost the same number. (But they can be slightly different even converged for large systems. But you don’t need to care it so much).  
NOTE: ehk:Hohenberg-Kohn energy, ehf: Harris-Faulkner energy.

“grep diffe lllmf” shows how the changed of total energy (and charges) during iteration. diffe mean changes of energy with previous iteration, (q) is for electron density difference as well. See also save.\* file, which only show ehk and ehk obtained by each iteration.

“ grep gap llmf” shows how the band gap changes (in the usual setting), two same numbers per iteration are shown now.

Thus we do have ground state energy. Although output of lmf is long, most of all are to monitor convergence (we will shrink it). As long as it converged well, you don’t need to look into it in detail. Eigenvalues are shown as

bndfp: kpt 1 of 4, k= 0.00000 0.00000 0.00000 ndimh = 122  
 -1.2755 -1.2008 -1.2008 -0.2052 -0.2052 -0.2052 -0.0766 -0.0766 -0.0766  
 -0.0174 -0.0174 -0.0174 0.1094 0.1095 0.1095 0.2864 0.2864 0.4170  
 0.4170 0.4736 0.6445 0.6445 0.6445

This is at k= 0.00000 0.00000 0.00000 . (because of historical reason, two same bndfp: are shown in each iteration; two band path method). “lmf cu| grep -A6 BZWTS” shows the Fermi energy (for insulator, we see band gap). Deep levels which gives little dependence on k are core like levels. These are in Ry; zero level is not so meaningful (for convenience, it is simply determined from the potential at MT boundaries).

rst.\* contains is the main output which contains electron density. mix.\* is a mixing file (which keeps iteration history). When you restart lmf again, it read rst.cu and mix.cu. If you start from lmfa result, please remove them. We can do parallel calculation with lmf-MPIK, we can invoke it with mpirun -np 8 lmf-MPIK cu. It should give the same answer.

## DOS, PDOS plot

We already have script to plot dos, band, and pdos from the result of lmf self-consistent calculations. We have scripts

jobUUUUUUtdos, jobUUUUUUband, jobUUUUUUpdos

. Look into these scripts, and then you see how to plot them.

For total DOS plot, it is better to check ctrl file; BZUUUUUUTETRA=1(this is default; thus make sure that BZUUUUUUTETRA do not exist or BZUUUUUUTETRA=1). In addition, it might be better to enlarge number of k point NKABC in ctrl file to have smooth curve. Then we do

jobUUUUUUtdos cu

This shows total DOS as

![DOS(Cu)](data:application/eps;base64,)

DOS(Cu)

The range of DOS and division of total DOS is given by DOSMAX and NPTS in ctrl. Edit tdos.cu.glt for gnulot for your presentation. Please look into job file in your bin directory. It is a small script.

For PDOS plot,

jobUUUUUUpdos cu

It shows figures (number of figures are number of atoms in the cell) in gnuplot (they are written in the same position on X-window; move top one a little). The command jobUUUUUUpdos is a little time-consuming because we use no symmetry to distinguish all lm channels. (PDOS is not yet implemented for SO=1 case; spin-orbit coupling $\L\dot S$ is added.) We can edit script of gnuplot (pdos.site\*.\*.glt) for your purpose. To plot again, run

gnuplot -persist pdos.site001.cu.glt

In principle, meanings of all data files are shown (see at the bottom of console output about lm ordering in a line), thus not so difficult to rewrite \*.glt. For example, to plot eg and t2g separately. (NOTE: site id is shown by lmchk).  
: Usually lmf and so on recognize options such as -v option. For example, ’ lmf gaas -vnspin=2’ or ’lmf gaas -vso=1’. This option changes values of variables defined in % const section. This is recorded in save.\* file, and also shown at the top of console output. However, joband so on, do not yet accept these options. Thus we need to modify ctrl file without using -v option. Or you need to write these option to these command by hand (we will fix this problem in future.)

## Band plot

For band plot, we have to set symmetry lines along which we plot eigenvalues. Collections syml.\* are in ecalj/MATERIALS/. (On Jan2017, we prepared automatic generator of syml.\* See Sec.[4.7.1](#sec:getsyml)). Choose and modify one of them and rename it. I will gather other samples soon. ’BZ wikipedia’ or something else will help you to interpret it.

To do band plot, we need syml.cu in your directory.

$ cp ~/ecalj/MATERIALS/Cu/syml.cu .

Then check syml.cu; it is

21 .5 .5 .5 0 0 0 L Gamma  
 21 0 0 0 1 0 0 Gamma X  
 0 (this is the terminator line)

We supply ten data for each lines (integer, three real, three real, two words). First line means, we calculate eigenvalues for **k** points from **k**=(0.5,0.5,0.5) to **k**=(0,0,0). "L Gamma" are names of two end points (.5 .5 .5) and (0 0 0) in this case. These names are used in a gnuplot script for band plot (bndplot.isp\*.glt). Second line means, we calculate eigenvalues for k points from **k**=(0,0,0) to **k**=(1,0,0). 3rd line means calculation just stop here. Units of **k** are in 2ALAT (or 2(ALAT+DALAT)@ if DALAT exist.). A line starting from ’#’ is neglected (comment line).

To do band plot, run

$ jobUUUUUUband cu

. This is for both nspin=1 and nspin=2 (These scripts try to determine the Fermi energy first. You may skip it in cases (but need to change the script)).

![band plot(Cu)](data:application/eps;base64,)

band plot(Cu)

### syml genearator

Jan2017, we added directory ecalj/GetSyml, which contains automatic generator of symmetry-line file syml.\* from ctrl.\*). Simultaneously, we plot a picture of BZ showing the symmetry lines by matplotlib. See ecalj/GetSyml/README. We use open softwares spglib and seekpath (these should be installed separetely). We have to add acknoledgement to them (see README) )when we use this generator.

## Useful samples: ecalj/MATERIAL/

Not only ecalj/lm7K/TestSamples (some of them are by older version), We have a material database in ecalj/MATERIALS/. Move to the directory, and type

$ ./jobUUUUUUmaterials.py

Then it shows a help. You see

...  
=== Materials in Materials.ctrls.database are:===  
 2hSiC 3cSiC 4hSiC AlAs AlN AlNzb AlP AlSb Bi2Te3 C  
 CdO CdS CdSe CdTe Ce Cu Fe GaAs GaAsUUUUUUso GaN  
 GaNzb GaP GaSb Ge HfO2 HgO HgS HgSe HgTe InAs  
 InN InNzb InP InSb LaGaO3 Li MgO MgS MgSe MgTe  
 Ni NiO PbS PbTe Si SiO2c Sn SrTiO3 SrVO3 YMn2  
 ZnO ZnS ZnSe ZnTe ZrO2 wCdS wZnS  
...

. For these simple materials (now 57 materials), input files can be generated, and run them automatically by a command ./jobUUUUUUmaterials.py below. The ctrls are stored in ecalj/MATERIALS/Materials.ctrls.database in a compact manner (in addition, options passed to ctrlgenM1.py and options to lmf-MPIK are included). See ecalj/MATERIALS/README about how to add new material to it; it is not difficult. The command ./jobUUUUUUmaterials.py gives ctrls.\* for these materials from descriptions in the Materials.ctrls.database. And then it generates ctrl file by calling ctrlgenM1.py internally, and run lmfa lmf-MPIK successively (when no –noexec).

Try ./jobUUUUUUmaterials.py Fe --noexec. (not fe but Fe as it shown above) at ecalj/MATERIALS/. Then it makes a directory Fe/ and set ctrl.fe (also ctrls.fe) in the directory. Without ’–noexec’, it does calculation for Fe successively. As for NiO and Fe, we see that ./jobUUUUUUmaterials.py gives SPECin generated ctrls and ctrl files. (Look into ctrls.fe; we need SPEC section when we add MMOM.)

Try jobUUUUUUmaterials.py GaAs Si.  
Then directories GaAs/ and Si/ are generated. See save.\* files containing total energies iteration by iteration. Starting from ctrl.\* in these directory, the command perform DFT calculations (Console output is stored in llmf, save.\* gives total energies. rst.\* contains self-consistent density, from which we can calculate energy bands and so on).

“./jobUUUUUUmaterials --all --noexec” generates ctrls and ctrl files of these materials. “./jobUUUUUUmaterials --all” do self-consistent LDA calculations for materials (it takes an hour or more. To change the number of cores for lmf-MPIK, set option -np (number of core). See help of ./jobUUUUUUmaterials (type this without arguments).  
To make band plot and so on for Fe, follow instructions already explained.

$ ./jobUUUUUUmaterials.py Fe (and need to type return)  
 (If you like start over, remove Fe/ under it first).  
 $ cd Fe  
 $ ./jobUUUUUUmaterials.py fe  
 (but it might be better to do --noexec, and observe Fe/ctrls.fe and  
 Fe/ctrl.fe first. grep conf llmf shows the initial electron distribution).  
 $ cat save.fe (this shows total energies of each iteration. 'c ' at  
 the first column gives converged result. 'h ' is from atm file.)  
 If it does not ends with 'c ...' line, something strange  
 occurs. see llmf (console out put of lmf is saved to llmf).  
 $ cp ../syml.fe .  
 $ jobUUUUUUband fe -np 4   
 (As I said, this shell script do not yet accept  
 options to lmf. Look into the script).  
 (This calculate fermi energy first for safe; it takes  
 some time)   
 $ jobUUUUUUtdos fe  
 $ jobUUUUUUpdos fe (as I said, this supress space-group symmetry, thus time consuming).

At the end of jobUUUUUUpdos, we show a help which pdos data is where (In pdos file, we have 26 numbers a line; first is energy, 2-26 are pdos for s,p,d,f,g; which is which are shown in the help). See joblmf file also (it contains options to invoke lmf. This is shown in save.\*. In principle, options in joblmf should be passed to band plot and so on. But not yet implemented (it is not so difficult; I have to do it).

After doing ./jobUUUUUUmaterials foobar, you may like move it back to original... In such a case, git works. At ecalj/, do

$ mv MATERIALS MATERIALS.bk  
 $ git checkout MATERIALS

Then you can see MATERIALS/ is moved back to just downloaded one.

## How to add spin-orbit coupling?

In the LDA level calculations, we can use  
SO=0 (no SO), SO=1(LdotS), or SO=2(LzSz) schemes.  
However, SO=1 is not the non-collinear method  
(z-direction is assumed to be spin direction; in cases  
we may need to primitive cell so that the z-direction  
is the spin direction).  
  
Within LDA, we have two possible ways.  
(A) Do LDA and/or QSGW with SO=0 first.  
Then apply the spin-orbit coupling by perturbation.  
or (B) do self-consitent LDA calculations with SO=1 (or =2).  
For semiconductors, we think you observe little differences.  
  
Let me explain the case (A).  
After converged with nspin=1, create new directory and copy  
 ctrl.gas, rst.gas, sigm.gas, QGpsi,  
to it. Then we set  
 nspin=2   
 METAL=3 (usually this is default)  
 SO=1 (this is ldots calculation off-diagonal elements included).  
 Q=band (we do not change potential.)  
in ctrl.gas.   
Then run  
>lmf gas >& llmfUUUUUUSO  
You can see "band gap with SO" by   
> grep gap llmfUUUUUUSO.  
Then you can see two same lines.  
 VBmax = 0.101949 CBmin = 0.236351 gap = 0.134402 Ry = 1.82786 eV  
 VBmax = 0.101949 CBmin = 0.236351 gap = 0.134402 Ry = 1.82786 eV  
(two lines per iteration is shown in metel mode).  
This is the band gap with SO as a first-order perturbation   
on top of the "QSGW without SO". When you use ctrl fil e generated by  
ctrlgenM1.py. You can do the above procedure with  
>lmf --rs=1,0 gas -vnit=1 -vso=1 -vnspin=2 -vmetal=3 --quit=band  
(--rs=1,0 read rst.gas but not write rst.gas. Run lmf --help.  
 The switch -v (-vso=1 in this case) replaces so=0 with so=1.   
 This is recorded in save.gas file).  
  
For band plot, you can use the same procedure   
for the case without SO. (Look into the shell script jobUUUUUUband.  
You have to modify it so that   
 '--rs=1,0 gas -vnit=1 -vso=1 -vnspin=2 -vmetal=3 --quit=band'   
is added to arguments for >lmf --band:syml ...).  
(--quit=band is not necessary if we like to renew eigenfunctions  
self-consistent. Anyway we expect little differences.)  
  
---  
  
\noindent QSGW with SO:\\  
For given sigm file, it is possible to attain self-consisency with SO=1(or=2) with keeping sigm (then we do not set Q=band).   
However, this imply that Vxc is fixed in QSGW,   
it is not necessary better than the above procedure.

## PROCASR (VASP format) generator

PROCASR mode for lmf (not yet in lmf-MPIK) Band weight decomposition. (Size of circles show the size of components. Superposed on band plot). See /MATERIALS/MgOUUUUUUPROCAR/README@, May19.2014

## efermi.lmf

This is generated by lmf(lmf-MPI), which is used in the job.

## Effective mass calculation

New band plot and effective mass calculation (curvature):  
(see sample at ecalj/MATERIALS/massUUUUUUfitUUUUUUtest/ and its README).  
------------  
We now read syml.\* in lm7K/fp/bndfp.F. Thus jobUUUUUUband is changed  
(nspin=1 or nspin=2 is automatically choosed by jobUUUUUUband command).  
We do not use plbnd anymore. If necessary, you can modify  
"writeband" subrouitne in lm7K/fp/bndfp.F by yourself.  
  
Follow steps to get effective mass...  
1.   
New syml read label of k point. It is shown by the gnuscript  
file "bandplot.glt". Type "gnuplot -p bandplot.isp1.glt" and so on,  
when you like to remake band plot.  
  
2.   
New syml allow a special input suitable to determine effective mass  
for semiconductor. An example of new syml is (this is a case of GaAs)  
------ syml example start ------------------  
#NOTE: ndiv2, ninit2 nend2 etolv etolc are for mass mode.  
#ndiv qleft(1:3) qright(1:3) llabel rlabel ndiv2 ninit2 nend2 etolv(Ry) etolc(Ry)  
5 0 0 0 .5 .5 .5 Gamma L 257 1 32 0.1 0.01  
5 0 0 0 1. 0 0 Gamma X   
5 0 0 0 .75 .75 0 Gamma K 257 1 32 0.1 0.01  
------- end ----------------------  
New feature is start from the next columns of symmetry points labels.  
In this example, 257 means a line connecting 0 0 0 (Gamma) and .5 .5 .5 (L) is divided  
into 256. Then we only calculate from the 1st point to the 32th point  
among 257 points.  
  
Then we see not only band\*.spin\*, but also  
we have files such as "Band001Syml001Spin1.dat"   
files which contains eigenvalues for selected bands, that is,  
we make a table of detailed plot of energy bands   
whose energy E at Gamma (exactly speaking, at  
left-end point), is evaltop-etolv (Ry) < E < econtop + etolc(Ry).  
  
(As its head line shows, Band\*Syml\*.dat file contains data  
 isyml,iq, ib,isp, QPE-EF, QPE-QPE(start), |q|, mass=2\*2\*(QPE-QPE(start))/|q|\*\*2).  
 here QPE(start) is QPE at the left-end of sym line.)  
  
This sample is only for Gamma-L and Gamma-K.  
  
Let us use this example for GaA and run jobUUUUUUband.  
(e.g., jobUUUUUUband gaas -np 2 -vnspin=2 -vso=1;   
 note that -vfoobar=xxx replace value of foobar with xxx in ctrl.gaas)  
Then you can see not only bandplot.isp1.glt.  
Then you can see usual energy bands plot in addition to the dense energy bands  
superposed on it. The dense energy bands are controlled by  
the part "257 1 32 0.1 0.01". In the current version, set all lines  
of etolv and etolc are the same.  
  
There are useful data, not only the eigenvalues in "Band001Syml\*Spin\*.mass"   
files. But, anyway, we have to determine effective mass from the eigenvalues  
on the dense symmetry mesh.  
  
  
3.   
Least square fit by gnuplot.   
  
We have an example is at ~/ecalj/MATERIALS/massUUUUUUfitUUUUUUtest  
Look into ./job and run it.   
See README in it.  
----  
For your convenience, we have dE/dk in the bnd\*.spin\* files.  
This is useful to determine the Fermi surface.  
See the efermi.lmf to read the Fermi energy.

## density and eigenfunction plot

We can generate density and $\sumUUUUUU{n} |{\bf k}n{{\bf k}n}({\bf r})|^2$ by --density mode (after 18mar2016) Usages are

>mpirun -np 24 lmf-MPIK si --density

This is for density. We have smrho.xsf, and rho1mt\*(1st components) and rho2mt\*(2nd components).

>mpirun -np 24 lmf-MPIK si --density,iq=1,ib=5

iq=1 means Gamma point. See index shown by the console output of lmf. The index ib is the band index.

>mpirun -np 24 lmf-MPIK si --density,iq=1,ib=5,6,7

In this case we have sum for ib=5,6,7.

Use xcrysden as

>xcrysden --xsf smrho.xsf

Then open menu ’ToolsDetaGrid’. Then plot isosurface and/or planes. For example, you need to set isovalue, and/or turn on ’display color plane’ buttons.

## LDA+U

There is a document of <file://../BACKUP/MarksOriginalDoc/fp.html#ldaplusu>. But I include all required things here. Usage is slighly changed. We have examples at  
 /ecalj/MATERIALS/erasldau,GdNldau@. I am going to add some samples.

To run LDA+U calculaiton, we need two extra steps in addition to usual LDA calculaitons:  
Step.(1) Set three tokens in ctrl file. LDA+U type, Parameters U and J.  
Step.(2) Prepare occnum.*ext* (initial condition of diagonal part of the density matrix).  
Let us explain one by one.  
  
For the point (1), we have a new line in SPECsections. In ecalj/MATERIALS/erasldau, it is

IDU= 0 0 2 2 UH= 0 0 0.1 0.632 JH= 0 0 0 0.055

The SPECtoken tells lmf that no U is to be added to the s or p channels, but that a U is to be added to the d and f channels. IDU=2 specifies LDA+U functional style 2; this is the "Fully Localized Limit" described in Liechtenstein, PRB 52, R5467 (1995)). IDU=1 specifies the "Around Mean Field" functional (Petukhov, PRB 67, 153106 (2003)). U=0.1 Ry is included on the d orbital, and U=0.632 is included on the f orbital. Additionally J=0.055 is put on the f orbital.

It is stored in .ext@. In principle, the density-matrix is read from and written to the file .ext@. Thus, in principle, we need a dmats.ext as an initial condition. However, we usually use diagonal-only density matric occunum.ext instead. If no .ext@, we read .eras@ instead. We explain it afterwards. Two density-matrices (for each spin) are written to this file in a (2l+1) by (2l+1) block for each l block for which a U is defined. dmats.ext is an ASCII file which you can read, and it’s quite useful to interpret what’s going on. The diagonal parts are the occupation numbers and are the most important. It is in the spherical harmonics (not the real harmonics). The harmonics is defined in the ecaljdetail.tex.

Note that in the LDA case, complete information is contained in the density, stored in file rst.ext. In the LDA+U, the Density-matrix is stored in file dmats.ext. Thus complete information is contained in the combination of rst.ext and dmats.ext.

The density matrix is printed out in the standard output. It can be represented on the real harmonics. To check it, run

>lmf eras --quit=dmat

. This stops promptly without side effect. Then the symmetrized density matrix represented on the spherical harmonics and real harmonics are shown (we make symmetrization for the given SYMOPS in ctrl file).

.(2)@  
Usually it is too tedious to supply dmats.ext itself. Instead, you can supply an "occupation numbers" file occnum.ext, which is a starting guess for the density-matrix (its diagonal part). occnum.ext has one line of (2l+1) numbers for the occupation numbers of the first spin, following by a line with the occupation numbers for the second spin. In the case of ecalj/MATERIALS/erasldau, we have occnum.eras.

Er has 11 f electrons, 7 of which go into the majority channel and 4 into the minority channel. There is some choice in which m states to fill and which to keep empty. A key point is that the self-consistent solution you end up with will depend on this choice. The ErAs test uses the following input file for occnum.eras :

0 0 0 0 0  
 0 0 0 0 0  
 1 1 1 1 1 1 1  
 0 1 1 1 1 0 0

The first and second lines are occupation numbers for the majority and minority d channel; the third corresponds to the majority f channel where all states are taken to be filled. The last line corresponds to the minority f channel. In this case, m=-2,-1,0,1 are filled and m=-3,2,3 are empty. As the script notes, different choices of starting occupation numbers lead to different self-consistent solutions. The one with the lowest energy is that which satisfies Hund’s rule (m=0,1,2,3 filled and m=-3,-2,-1 empty).

It is possible to use ’real harmonics (see ecalj/fpgw/exec/jobpdos for definition)’ instead of ’spherical harmonics’. As for occnum.ext, you can use

% real  
 0 0 0 0 0  
 0 0 0 0 0  
 1 1 1 1 1 1 1  
 0 1 1 1 1 0 0

. Here '% real' means on the basis of real harmonics. Check whether this is correctly supplied or not by >lmf –quit=dmat eras@.

To run with spin-orbit coupling, set HAM=1, then run >grep IORBTM -A20 llmf , here llmf is the standard output of lmf. Then you see orbital moments at ’IORBTM’.

NOTE for gwsc:  
When you just like to generate initial condition for gwsc by LDA+U, you have to exit right after sigm generated (add right after ix ’iteration over==’@ in gwsc). Then gwsc stops right after sigm generated. Then you have to remove or comment out IDU (or set all zero) in ctrl file. Then run lmf together with obtained sigm file. This is because sigm contains LDA+U kind of effect. After this, check energy band (jobband) for check. And run usual gwsc— then you can perform QSGW with the LDA+U’s initial condition.

## QSGW after LDA+U

We modified 7K/subs/mUUUUUUrdctrl.F@(feb2019), so that An option which makes “QSGW after LDA+U” easy to run. A line for atom section

IDU=0 0 0 12 UH=0 0 0 .7 JH=0 0 0 0

works in ctrl file. Here 12=10+2 instead of 2 means that we set UH=JH=0 when sigm.\* exits. Thus we can perform QSGW(without U term) calculaiton, after LDA+U calculaiton. This can be convenient for cases when we control the initial condition of QSGW.

## partially occupied core-hole

See <file://../BACKUP/MarksOriginalDoc/fp.html#ldaplusu> around. Search the keyword ldau in fp/lmfp.F sudmtu.F which read occnum.gdn.

Core-hole procedure is one of the important procedure. An example setting for Nd case is

--------------------------  
 ATOM=Nd Z=60 R=3.00  
 C-HOLE=4f C-HQ=-11 P=6.5,6.5,5.5,5.2　　<-- this line is added  
--------------------------

1. C-HOLE=4f means that 4f core is to set “core hole”.
2. P=6.5,6.5,5.5,5.2 is to set principle (fractinal) quantum number explicitly for valence electrons. 　To set this, try lmfa nd|grep conf first. The you can see the electronic configulation of an atom. Key point here is to set “5.2” for f channel. Thus 4f is treated as core and 5f as valence. Fractional parts (0.2 of 5.2) means the radial funciton contains nodes as 5f and closer to the energy right on the change of 4f to 5f. If 5.9, it is closer to the enrgy almost have noded of 6f.
3. C-HQ means that corehole is −１１. Thus we have only 14-11=3 electrons for 4f (spherically occupied).
4. Run lmfa. If it does not show that (warning) atom not neutral@, it is the neutral atom. You can also chage the message tagged by core hole:@ Simultaneously you can control initial total valence charge by Q=...

Very recently (aug2019), I fixed a bug of EFERMI(fermi energy for GW calculaiton, given by /main/heftet.m.F@) for the case C-HOLE is supplied.

# How to run QSGW calculation?

In the QSGW, we calculate a *non-local exchange-correlation potential* $V^{\rm xc}({\bf r},{\bf r}')$, by a procedure of GW calculation (very time-consuming part). Then difference $V^{\rm xc}({\bf r},{\bf r}')V-VUUUUUU{\rm xc}^{\rm
LDA}({\bf r})\delta({\bf r}-{\bf r}')$ is stored into sigm.\* file. The potential file is a key to perform QSGW calculations as seen in Fig.[[gwscpicture]](#gwscpicture). The contains static non-local potential $\SigmaUUUUUU{\rm QSGW}-VUUUUUU{\rm xc}^{\rm LDA}$.

Then, we again do one-body calculation by lmf (or lmf-MPIK) where we add this sigm to one-body potential ; when we run or -MPIK@(k-parallel mpi version), is read and added to the one-body potential if we have HAM=12 in ctrl.\*. Thus this means that we replace $VUUUUUU{\rm xc}^{\rm LDA}({\bf r})\delta({\bf r}-{\bf r}')$ with $V^{\rm xc}({\bf r},{\bf r}')$.  
This iteration cycle is performed by a script “gwsc” as we explain later on. (In the default setting of ctrl.\* file, lmf try to read sigm.\* file as long as it exists. If not, do lmf or lmf-MPIK calculation.

To start QSGW calculation by gwsc, we need not only ctrl.si, but also another input file GWinput. Its template GWinput.tmp can be generated by the Step.2 as follows.

As a summary, you have to follow steps below in order to perform QSGW calculation.

1. Perform from the Step.1 thru the Step.4 (up to lmfa) in Sec.[4.5](#lm7K-scf). (as same as the case of LDA/GGA).

* You don’t need to perform LDA calculation in advance, since gwsc perform LDA/GGA calculation at its beginning. (It means that we start from the one-body Hamiltonian in LDA/GGA as initial condition.In cases, LDA/GGA give poor initial conditions for QSGW; in such a case, we may need another trick to prepare starting point.).
* [Caution: We have to use the same ( in the expansion of eigenfunctions in each MT) for all the MT spheres. (This is due to historical reason; we may need to fix this.)] xxx need to explain xxx
* The purpose of this script is to get GWinput.tmp. Other files generated are not used in the following stage.

1. .  
   GWinput is the input file describing the computational conditions for calculation. Usually, the default setting gives reasonable results. To reduce computational time, we may use (atom)=2@ for oxygen sites (this may be also other small atoms.). lcut GWinput -A1@ shows for each atomic sites) These step 2. and step 3. are just only to get GWinput.

## GWinput

In order to perform QSGW, one another input file GWinput (no extension) is necessary in addition to ctrl.\*. Thus all input files for QSGW is just two files, ctrl.\* and GWinput. A template GWinput can be generated by a script mkGWINUUUUUUlmf2. You may have to modify it in cases for your purpose.  
Let us start from ctrls.si;

#id = Si  
%const bohr=0.529177 a= 5.43095/bohr  
STRUC  
 ALAT={a}   
 PLAT=0 0.5 0.5 0.5 0 0.5 0.5 0.5 0   
SITE  
 ATOM=Si POS=0.0 0.0 0.0  
 ATOM=Si POS=0.25 0.25 0.25

. Do ctrlgenM1.py si --tratio==1.0 --nk1=6 and copy ctrlgenM1.ctrl.si to ctrl.si. NOTE: the option –tratio=1.0@ means we use touching MT; this can be checked by lmchk si; since defaults is almost unity (–tratio=0.97@), this is irrelevant, just to explain options.

We have to write GWinput. The default is given automatically by a command mkGWINUUUUUUlmf2;

$ lmfa si (lmfa is needed to do in advance).  
 $ mkGWINUUUUUUlmf2 si  
 ......  
 == Type three integers n1 n2 n3 for Brillowin Zone meshing for GW! ==  
 n1=

Then it pause and ask numbers. You have to type three numbers as 2+ return + 2+return+2 return.

== Type three integers n1 n2 n3 for Brillowin Zone meshing for GW! ==  
 n1= 2  
 n2= 2  
 n3= 2  
 2 2 2  
 ...(skip)...  
 OK! GWinput.tmp is generated!

Generated file is GWinput.tmp; you have to copy it to GWinput.

$ cp GWinput.tmp GWinput

These ’2 2 2’ you typed is reflected in a section ’n1n2n3 2 2 2 ’ in GWinput. This means 2x2x2 (8 points in 1st BZ). You can edit it, and change it to e.g. ’n1n2n3 4 4 4’ if you like to calculate self-energy on dense BZ mesh 8x8x8.

The template of GWinput is usually not so bad. But it may give a little expensive setting (or not very good enough in cases).

NOTE: GWinput allows some options. Some newer points are

1. For the atoms without 3d involved, you can use lcutmx=2. Typically we use lcutmx=2 for oxygen.
2. 0 0.0001@ given in GWinput is effective to stabilize the convergence for metals.

## Run gwsc script

Let us perform QSGW calculation. For this purpose, we use a script gwsc. We need to do lmfa in advance. Then do (not need to do lmf);

gwsc (number of iteration+1) -np (number of nodes) (id of ctrl)

If (number of iteration+1)=0, it gives one-shot calculation from LDA. But it is different from the usual one-shot; since it calculates off-diagonal elements of self-energy also, we can plot energy band dispersion. In cases (for usual semiconductors), it can give rather reasonable results in comparison with experiments from practical point of view.

This is an example of one iteration of QSGW cycle. (now a little different but essentially similar)

takao@TT4:~/ecalj/test1$ gwsc 0 -np 2 si  
gwsc 0 -np 2 si  
### START gwsc: ITER= 0, MPI size= 2, TARGET= si  
--- No sigm nor sigm.$TARGET files for starting ---  
 ---- goto sc calculation with given sigma-vxc --- ix=,0  
No sigm ---> LDA caculation for eigenfunctions   
 Start mpirun -np 2 /home/takao/ecalj/TestInstall/bin/lmf-MPIK si > llmfUUUUUUlda   
OK! --> Start echo 0| /home/takao/ecalj/TestInstall/bin/lmfgw si > llmfgw00   
OK! --> Start echo 1|/home/takao/ecalj/TestInstall/bin/qg4gw > lqg4gw   
OK! --> Start echo 1|mpirun -np 2 /home/takao/ecalj/TestInstall/bin/lmfgw-MPIK si> llmfgw01   
OK! --> Start /home/takao/ecalj/TestInstall/bin/lmf2gw >llmf2gw  
OK! --> Start echo 0|/home/takao/ecalj/TestInstall/bin/rdata4gwUUUUUUv2 > lrdata4gwUUUUUUv2   
OK! --> Start echo 1| /home/takao/ecalj/TestInstall/bin/heftet > leftet   
OK! --> Start echo 1| /home/takao/ecalj/TestInstall/bin/hchknw > lchknw   
OK! --> Start echo 3| /home/takao/ecalj/TestInstall/bin/hbasfp0 > lbasC   
OK! --> Start echo 3| mpirun -np 2 /home/takao/ecalj/TestInstall/bin/hvccfp0 > lvccC   
OK! --> Start echo 3| mpirun -np 2 /home/takao/ecalj/TestInstall/bin/hsfp0UUUUUUsc > lsxC   
OK! --> Start echo 0|/home/takao/ecalj/TestInstall/bin/hbasfp0 > lbas   
OK! --> Start echo 0| mpirun -np 2 /home/takao/ecalj/TestInstall/bin/hvccfp0 > lvcc   
OK! --> Start echo 1| mpirun -np 2 /home/takao/ecalj/TestInstall/bin/hsfp0UUUUUUsc > lsx   
OK! --> Start echo 11| mpirun -np 2 /home/takao/ecalj/TestInstall/bin/hx0fp0UUUUUUsc > lx0   
OK! --> Start echo 2| mpirun -np 2 /home/takao/ecalj/TestInstall/bin/hsfp0UUUUUUsc > lsc   
OK! --> Start echo 0| /home/takao/ecalj/TestInstall/bin/hqpeUUUUUUsc > lqpe   
OK! --> == 0 iteration over ==  
OK! --> Start mpirun -np 2 /home/takao/ecalj/TestInstall/bin/lmf-MPIK si > llmfUUUUUUgwscend.0   
OK! ==== All calclation finished for gwsc 0 -np 2 si ====

Here echo (integer) is readin in at the beginning of the code. To see it, please look into gwsc script (gwsc is at ecalj/fpgw/exec/ and copied to your bin/ by make install2). In anyway, this console output shows calculations finished normally.

Now we get rst.si and sigm.si file which contains (static version of) self-energy minims $VUUUUUU{\rm xc}^{\rm LDA}$. What we did is the one-shot GW from LDA result; but note that we calculate not only diagonal elements but also off-diagonal elements.

We can write energy dispersion (band plot) in the same manner in LDA. To do it, we need rst.si, sigm.si, ctrl.si, QGpsi. (but QGpsi is quickly reproduced). After you have syml.si (e.g. in ecalj/MATERIALS/), Do

$ jobUUUUUUband si

![Si, one-shot GW with off-diagonal elements](data:application/eps;base64,)

Si, one-shot GW with off-diagonal elements

[sigwscone]

You can observe large band gap as shown in the Fig.[[sigwscone]](#sigwscone). (To see it again, gnuplot bnds.gnu.si -p. All plots are in gnuplot, thus it is easy to replot it as you like).

We have QPU file (and also QPD for spin=2), which contains content of the diagonal part of self-energy. It will be explained elsewhere.

You can make total DOS and PDOS plot by

$ jobUUUUUUtdos si  
 $ jobUUUUUUpdos si

CAUTION:pdos plot is not allowed for so=1. (even tdos–> ask to t.kotani.)  
To get final QSGW results, we have to repeat iteration until eigenvalues are converged. Note that total energy shown by console output llmf (and also shown in save file) is not so meaningful in the QSGW; we just take it as an indicator to check convergence. Let us repeat 5 iteration more. "-np 2" means one core to use.

$ gwsc 5 -np 2 si  
### START gwsc: ITER= 5, MPI size= 2, TARGET= si  
 --- sigm is used. sigm.$TARGET is softlink to it ---  
 ---- goto sc calculation with given sigma-vxc --- ix=,0  
 we have sigm already, skip iter=0  
 ---- goto sc calculation with given sigma-vxc --- ix=,1  
 ...(skeip here) ...  
  
OK! --> == 5 iteration over ==  
OK! --> Start mpirun -np 2 /home/takao/ecalj/TestInstall/bin/lmf-MPIK si > llmfUUUUUUgwscend.0   
OK! ==== All calclation finished for gwsc 0 -np 2 si ====

Note that we do skip 0th iteration (it is for one-shot from LDA) since we start from rst.si and sigm.si given by one-shot LDA. Thus we do just five iterations. Information of eigenvalues are in QPU.(number)run files. (for magnetic systems with nspin=2), wee have QPD.(number)run also). Check it by ls;

$ ls QPU.\*run  
 QPU0.run QPU.1run QPU.2run QPU.3run QPU.4run QPU.5run

(These are overwritten when we again repeat gwsc; be careful.) Note that QPU0.run was old one when you did 1-shot GW from LDA at the beginning. In anyway \*.0run are confusing files; remove them).

In order to check convergence calculations going well during iteration, do

$ grep gap llmf\*

This shows how band gap changes in llmf.\*run files. In metal cases, we need to compare QPU file, magnetic moment or grep '[xc] save.\*; this shows end of lmf iteration. Energy is not so meaningful but can be indicator to convergence.

Let us check convergence of the QSGW calculations. For this purpose, it is convenient to take a difference of QPU(QPD) files by a script dqpu. These files are human readable. To compare QPU4.run and QPU5.run, do

$ dqpu QPU.3run QPU.4run

Then we see a list of numbers (these are the differences of values in QPU files). Then it shows at the bottom as

Error! Difference>2e-2 between: QPU.4run and QPU.5run   
 : sum(abs(QPU-QPD))= 0.05736

but you don’t need to care it so much. You rather need to check the difference of values. I can say most of all difference (especially around the Fermi energy are ) are almost 0.00eV or 0.01eV, we can judge QPEs are converged. If not converged well, you may need to repeat gwsc again. (when the size of two QPU files are different, dqpu stops.)

![band plot(Si, QSGW one-shot test) and band(Si) (GGA)](data:application/eps;base64,) ![band plot(Si, QSGW one-shot test) and band(Si) (GGA)](data:application/eps;base64,)

![band(GaAs), QSGW (test case)](data:application/eps;base64,)

band(GaAs), QSGW (test case)

## Spectrum function: How to calculate $\langle {\bf q}n|\Sigma(\omega)|{\bf q}n\rangle$

We have an example at ecalj/MATERIALS/SiSigma/, where you can just type job. It calls a shell script gwsigma, which is just a modification of gwsc for spectrum function plotting. If you have sigm.\*, it will automatically read it as in the case of gwsc.

By the script gwsigma, we calculate the diagonal elements $\rangle \psi({\bf q},n)|\sigmaUUUUUU{\rm c}(\omega) |\psi({\bf q},n)\langle$. Thus we need to set ${\bf q}$ and band index for which we calcualte. In addition, we need to set resolution of .

1. Set <QPNT> section. This section is to set the q point, and band index for which we calculate the self energy. In addition, energy mesh for plotting is set.

* (A) Set q point set   
    
   If you set   
   ----------  
   \*\*\* all q -->1, otherwise 0; up only -->1, otherwise 0  
   1 0  
   ----------  
   You will have self-energy for all irreducible k points. This may be needed for A(omega).  
   or   
   You have to set all q points as  
   ----------  
   \*\*\* q-points, which shoud be in qbz.,See KPNTin1BZ.  
   3 <--- number of readin q point   
   1 0.0000000000000000 0.0000000000000000 0.0000000000000000 <--1st number is irrelevant  
   2 -0.5000000000000000 0.5000000000000000 0.5000000000000000  
   3 0.0000000000000000 0.0000000000000000 1.0000000000000000  
   ----------  
   To know allowed q points on regular mesh point, run the command "mkGWINUUUUUUlmf2", then  
   supply n1,n2,n3. The templete of GWinput.tmp contains all possible q points. Edit it.  
     
   NOTE:Anyq option can allow you to specify any q points by shifted mesh technique.  
   (if necessary, but only for some special purpose).  
    
  (B) Band index set  
   It is specified by the section  
   ----------  
   \*\*\* no. states and band index for calculation.  
   2  
   4 5  
   ----------  
   means the self-energy for band index 4 and 5. Just two bands.  
   If you like to plot self energy from 1 through 8, use  
   \*\*\* no. states and band index for calculation.  
   8  
   1 2 3 4 5 6 7 8  
   ----------  
   If you need 17 bands for example, it should be 1 2 3 4 5 6 7 8 9 10 11 12 13 14 15 16 17  
   in addition to 17 (number of bands at the first line).   
    
    
  (C) energy mesh set  
   At the bottom of <QPNT> section, we have  
   ----------------  
   \*\*\*\*\*  
   0.01 2.0  
   ----------------  
   Two real number should be supplied.  
   These are dwplot and omegamaxin, read in hsfp0.m.F by a line   
   read (ifqpnt,\*,err=2038,end=2038) dwplot,omegamaxin  
   dwplot (=0.01 Ry) is mesh for self energy.  
   omegamaxin=(2.0 Ry) means the range "-2 Ry to 2 Ry" for self-energy plot.  
    
   Note that imaginary part of Sigma is given as the comvolution of ImW(omega) and the pole of Green's function  
   (esmr in GWinput gives energy smearing of the pole). Resolution for Im W (near omega=0) is by dw in GWinput.)  
   I think that the reolution of self-energy is ~ 0.05 eV in the default setting.  
   This is because {\tt dw} \sim {\tt esmr} \sim 0.05 eV.

1. Run gwsigma. This will run   
    echo 4| mpirun -np 24 hsfp0,  
    after dielectric funcition is calculated.  
    Then we have SEComg.UP (DN) files, Look for file handle, ifoutsec,  
    for the file in fpgw/main/hsfp0.m.F to see format for the file.   
    (not hsfp0.sc.m.F but hsfp0.m.F). Search a line  
    open(ifoutsec,file='SEComg'//sss) (around hsfp0.m.F L1052)  
    You can find that we use folloing lines to plot SEComg.\*.  
    ----------------   
    write(ifoutsec,"(4i5,3f10.6,3x,f10.6,2x,f16.8,x,3f16.8)")  
    & iw,itq(i),ip,is, q(1:3,ip), eqx(i,ip,is),  
    & (omega(i,iw)-ef)\*rydberg(), hartree\*zsec(iw,i,ip) !,sumimg ----------------   
    This means we use energy in eV.   
    iw: omega index  
    itq(iq): band index specified by <QPNT>  
    ip: k point index specified by <QPNT>  
    is: spin index  
    q: q vector (cartesian in 2pi/alat)  
    eqx: eigenvalue in eV. (I think relative to the Fermi energy)  
    (omega(i,iw)-ef)\*rydberg(): omega relative to the Fermi energy  
    hartree\*zsec(iw,i,ip): Self energy. real and imaginary part.(complex, two values)  
     
    You can only repeat echo 4| mpirun -np 24 hsfp0   
    when you change setting in <QPNT> section.
2. Example. There is an example MATERIALS/SiSigma/

* plot 'SEComg.UP' u ($9):($10) w l,'' u ($9):($11) w l  
   can give a plot for Re (SigmaUUUUUUc(omega)) and Im(SigmaUUUUUUc).  
    
   9th: energy in eV (omega(i,iw)-ef)\*rydberg()  
   10th: real part Re hartree\*zsec(iw,i,ip)   
   11th: imag part Im hartree\*zsec(iw,i,ip)

1. To get integrated spectrum function (DOS), we need to superpose all the spectrum function (All q points and all band index). Be careful about the degeneracy (multiplicity) for each q points. You have to build it from SEComg file. To know the multiplicity, search following lines ofkeyword Multiplicity in the console output of qg4gw (lqg4gw).

* Anyway, consider about “is it worth to do?” To confirm your result, use sum rule (sum of spectrum weight). And pay attention to the relation between real and imag parts (Hilbert transformation).

# gwsc script to perform QSGW

## outputs of gwsc

![Shell script gwsc to perform QSGWcontains an iteration loop to make sigm (and eigenvalues, eigenfunctions) self-consistent. The iteration loop is written in ecalj/fpgw/exec/gwsc (a bash script). Exactly speaking, we have to pass all the required data (not only eigenvalues and eigenfunctions, but also crystal structure, q+G vectors, symmetry information and so on) to GW part. The purpose of the GW part is to calculate \SigmaUUUUUU{\rm QSGW}-VUUUUUU{\rm xc}^{\rm LDA}.](data:application/eps;base64,)

Shell script gwsc to perform QSGWcontains an iteration loop to make sigm (and eigenvalues, eigenfunctions) self-consistent. The iteration loop is written in ecalj/fpgw/exec/gwsc (a bash script). Exactly speaking, we have to pass all the required data (not only eigenvalues and eigenfunctions, but also crystal structure, **q+G** vectors, symmetry information and so on) to GW part. The purpose of the GW part is to calculate $\SigmaUUUUUU{\rm QSGW}-VUUUUUU{\rm xc}^{\rm LDA}$.

When gwsc runs normally, it gives console output as follows. This is a case for ctrl.gaas for  
>gwsc 10 -np 24 gaas@  
. Without arguments, typing gwsc shows usage as  
>An example of usage: gwsc 5 -np 4 si, where 5 means 5+1 iterations@  
. We recommend you do look into the script gwsc. It uses run, which is a special subroutine of bash; but not so difficult to understand it. (In the followings, we assume /home/binx/ is your bin directory at which we have all binaries for ecalj.)

### START gwsc: ITER= 10, MPI size= 24, TARGET= gaas  
--- No sigm nor sigm.$TARGET files for starting ---  
 ---- goto sc calculation for given sigma-vxc --- ix=,0  
No sigm ---> LDA caculation for eigenfunctions   
OK! --> Start echo --- | mpirun -np 24 /home/binx/lmf-MPIK gaas > llmfUUUUUUlda   
OK! --> Start echo 0 | /home/binx/lmfgw gaas > llmfgw00   
OK! --> Start echo 1 | /home/binx/qg4gw > lqg4gw   
OK! --> Start echo 1 | mpirun -np 24 /home/binx/lmfgw-MPIK gaas > llmfgw01   
OK! --> Start echo --- | /home/binx/lmf2gw > llmf2gw   
 ... (preparation stage ends here; start main stage) ...  
OK! --> Start echo 0 | /home/binx/rdata4gwUUUUUUv2 > lrdata4gwUUUUUUv2   
OK! --> Start echo 1 | /home/binx/heftet > leftet   
OK! --> Start echo 1 | /home/binx/hchknw > lchknw   
OK! --> Start echo 3 | /home/binx/hbasfp0 > lbasC   
OK! --> Start echo 3 | mpirun -np 24 /home/binx/hvccfp0 > lvccC   
OK! --> Start echo 3 | mpirun -np 24 /home/binx/hsfp0UUUUUUsc > lsxC   
OK! --> Start echo 0 | /home/binx/hbasfp0 > lbas   
OK! --> Start echo 0 | mpirun -np 24 /home/binx/hvccfp0 > lvcc   
OK! --> Start echo 1 | mpirun -np 24 /home/binx/hsfp0UUUUUUsc > lsx   
OK! --> Start echo 11 | mpirun -np 24 /home/binx/hx0fp0UUUUUUsc > lx0   
OK! --> Start echo 2 | mpirun -np 24 /home/binx/hsfp0UUUUUUsc > lsc   
OK! --> Start echo 0 | /home/binx/hqpeUUUUUUsc > lqpe   
OK! --> Start echo --- | mpirun -np 24 /home/binx/lmf-MPIK gaas > llmfUUUUUUoneshot   
 ... (this is the end of main stage) ...  
== 0 iteration over ==  
 ---- goto sc calculation for given sigma-vxc --- ix=,1  
OK! --> Start echo --- | mpirun -np 24 /home/binx/lmf-MPIK gaas > llmf   
OK! --> Start echo 0 | /home/binx/lmfgw gaas > llmfgw00   
 ... (lines here omitted ) ...  
OK! --> Start echo 0 | /home/binx/hqpeUUUUUUsc > lqpe   
== 1 iteration over ==  
 ... (lines here omitted ). ..  
== 2 iteration over ==  
 ... (this repeat until ITER= 10(the first argument to gwsc) ) ...

This shows that gwsc invoke lmf-MPIK,lmfgw,qg4gw,... successively. echo 3hbasfp0 means running a fortran program hbasfp0 with the argument ’3’ from the standard input (read(\*,\*) in fortran code). We can divide these successive calls to “preparation stage” and “main stage”. Preparation stage is just to prepare eigenfunctions and so on which are required for the “main stage” of GW calculation. At the end of “main stage”, we have the potential file sigm.

As it shows, console output are going to l\* files.

## **Preparation stage of gwsc**

At the end of this stage, we get required eigenfunctions, BZmesh data, and so on, which are required for the main stage. Note that **echo 0 lmfgw** means supply an integer to the fortran program **lmfgw** from standard input by read(\*,\*).

* lmf-MPIK (k-parallel version of lmf)  
  This is the one-body calculation for given sigm.gaas. At the beginning, we do not have sigm.gaas. In this case lmf-MPIK just perform LDA/GGA calculation.
* **echo 0 lmfgw**:  
  Get some small information files to start **qg4gw**. If you type just lmfgw, and observe what occurs. It shows a menu and pauses (asking you to supply an integer); then we supply 0 in this case. (If we do **echo 0 lmfgw**, no pause occurs.)
* **echo 1 qg4gw** : Get ${\bf k}$ points used in the calculations and the corresponding ${\bf G}$ vectors. And what is the irreducible k point (See console output of qg4gw. gwsc keeps it in lqg4gw).  
  Since we use the offset-Gamma method for BZ integration for , we need shifted mesh points to calculate at offset-Gamma points. The ${\bf q}$ vectors of offset-Gamma method is in Q0P file. (If you have two points in Q0P, we see two shifted mesh points in addition to regular mesh points.) Remember that cutoff of ${\bf G}$ is given by QpGcutand QpGcut in GWinput. (Based on the experiences, we use smaller QpGcut to reduce computational time. Explained in other section xxx).
* **echo 1 lmfgw-MPIK** : Calculate the LDA eigenfunctions, eigenvalues, and $\langle \psi | V^{\rm LDA}UUUUUU{\rm xc}| \psi \rangle$ at the irreducible ${\bf k}$ points (shown at the bottom of output lqg4gw of qg4gw.)
* **lmf2gw**: store these data into DATA4GW2 and CphiGeig, whose I/O is controlled by a key subroutine **gwinput2.f**.

## **Main stage of gwsc**

We can start the main stage of calculation from these files;  
DATA4GWUUUUUUV2 CphiGeig QGpsi QGcou Q0P QIBZ SYMOPS BZDATA HAMindex CLASS@;  
these files contains eigenfunctions and so on in the manner of Eq.(17) of , eigenvalues and other required information. This is the starting point of the GW calculation.

: computational conditions.

: Crystal structures and so.

: Eigenvalues and Eigenfunctions

: q and G vector for the eigenfunctions(q means **k** in the previous section),

: q and G vector for the Coulomb matrix

: q points near q=0 instead of q=0 (offset Gamma points)

: irreducible q points (This is also contained in BZDATA).

: point group operation

: q points date (and tetrahedron weights if necessary) for BZ integrals.

: Hamiltonian index, all required complex index for Hamiltonian of PMT method. (See the top of subroutine write in lm7K/subs/m.F. This is also in fpgw/gwsrc/m.F. Identical files are in two different directory— it should be avoided in future.)

: class information or atomic sites (equivalent sites).  
With these files, we do the main stage as

* **rdata4gw2**: Read DATA4GW2 and so on, and decompose it into files required in the followings. And calculate PPOVL\* files (overlap matrix of interstitial plane waves. Because of technical reasons some different types of PPOVL\* with **q**-point index).
* **heftet** : Get the Fermi energy EFERMI by tetrahedron method. It is used in **hx0fp0**.
* **hchknw** : stores the number of required points along real-axis into NW.  
  (NW is not essentially used, but is supposed to exist in the followings.)
* **echo 3hbasfp0**: gives the product basis for Core exchange.
* **echo 0hvccfp0**: gives the Coulomb matrix for the Core exchange.
* **echo 3hsfp0**: gives the Core exchange part of the self-energy. [[12]](#footnote-78)
* **echo 0hbasfp0**: gives the product basis.
* **echo 0hvccfp0**: gives the Coulomb matrix .
* **echo 1hsfp0**: gives the exchange part of the self-energy.
* **echo 11hx0fp0**: gives the correlated part of the screened Coulomb interaction .
* **echo 2hsfp0**: gives the correlated part of the self-energy.
* **echo 0hqpe**: gather data and write down final results into sigm, QPU, TOTE.UP files.

## **Other functions (or scripts)**

In addition to **gw**, there are some other additional scripts and functions.

* **gw** : The one-shot calculation. Lifetime(impact ionization rate) of QPs.
* **gwsc** : QSGW calculation explained here
* **epsPP, eps** : Dielectric function without or with local-field effects.
* run-mode 4 of **hsfp0**: to plot the spectrum function .(need to be fixed again probably).
* **epsPP** : non-interacting spin susceptibility. One-degree of freedom like Rigid moment approx. After it ends, you need to do and/or to get the full spin susceptibility.
* **genMLWF** : Wannier funciton and its matrix elements of the Screened Coulomb interaction.

# Usage problems, QandA error messages.

1.Bandplot for FSMOMMETHOD/=0   
 Even when you use FSMMOMMETHOD/=0 in GWinput for gwsc,   
 you need to set FSMOMMETHOD=0 (or comment it out) when you run jobUUUUUUbandUUUUUUnspin2.  
 [If you run jobUUUUUUbandUUUUUUnspin2 with FSMOMMETHOD/=0, it make a shift   
 (adding bias magnetic field).]  
  
2.Note that ctrlgenM1.py automatically set this for --systype=molecule.  
 Then we have   
 TETRA=0  
 N=-1 #Negative is the Fermi distribution function W= gives temperature.  
 W=0.001 #W=0.001 corresponds to T=157K as shown in console  
 In addiiton, FSMOM (nUUUUUUup-nUUUUUUdown) is needed (FSMOMMETHOD=1)if we  
 have magnetic moment.  
  
3. core>evalence message.  
 Ecore is grater than Evalence.  
 For save, we do not allow this.  
 Complare ECORE file and valence levels, shown in log file or  
 console output.  
  
4. If you see a error message from lmf (e.g., internally called in the gwsc script).;  
 Exit -1 rdsigm: Bloch sum deviates more than allowed tolerance (tol=5e-6)  
 You have to enlarge RSRNG so that lmf finsh normally.  
  
5. Back ground charge and fractional Z.  
 You can use fractional numbers for ATOMUUUUUUZ, and also can set  
 valence charge by BZUUUUUUZBAK (I removed BZUUUUUUVAL).  
 You see console out put, e.g,  
 "Charges: valence 19.80000 cores 8.00000 nucleii -28.00000  
 hom background .20000 deviation from neutrality: 0.00000  
 . This is a case with BZUUUUUUZBAK=.2.  
  
 NOTE: at the first iteration, Charges: shows such as  
 Charges: valence 8.00000 cores 20.00000 nucleii -28.00000  
 hom background 0.12300 deviation from neutrality: 0.12300  
 because of the initial condition by superposition of atoms. It show  
 deviation seems nonzero. But charge should be conserved from the  
 next iteration.  
  
6. Not converged in metal. --->mixing may help  
 For example, if you try metal such as Bi2Sr2CuO6, it may fail at LDA/GGA level.  
 Then use ITER MIX=A2,b=.2. or something (.2 means it only mix 20% of output to give  
 new input for next iteration). Then I see convergence. (b is the  
 mixing parameter.  
  
7. Use PZ or not.  
 If spillout of core is not so small (more then 0.05 or something.),  
 it is better to use PZ(lo). Treat the core as valecne.  
 Bi4d is such a case. Maybe use PZ=0,0,4.9  
  
8. Core treatment   
 See 10.1103/PhysRevB.76.165106 (Eq.35 and after).  
 Now I usually not use CORE2 (CORE1 only).  
  
9. ERROR EXIT! rgwina: 2nd wrong l valence  
 This may be because you use wrong GWinput.  
 Back it up. And run mkGWINUUUUUUlmf2 (any n1 n2 n3 is fine).  
  
10. Known bug.  
 Error occurs when system is anisotroic such as CuAlTe2.   
 Temporary fix is "Add token NPWPAD=100 in HAM category".  
 (guess of used APW fails (more than expected)).  
 CuAlTe2,CuGaTe2 cases.  
  
11. Known bug  
 a little unstable when metal GGA, especially when we have large empty regions.

# Cautions for usage

1. == not meaningful total energy in QSGW===  
   Total energy shown in QSGW mode in current version is not meaningful. (just treat as an indicator to convergence).
2. == Do we use VWN or GGA for QSGW? ===  
   In principle, QSGW results should not depends on VWN or GGA (XCFUN=1 or 103 in ctrl). But there is minor dependence, because  
   1. frozen core density.  
   2. core eigenfunctions.  
   3. radial basis functions  
   4. Slight numerical reason  
   (This is probably because Sigma-interpolation procedure But not exactly figured out yet affect about 0.02eV as for band gap for GaAs. ). In anyway, use VWN (HAM=1) as standard. And such technical things affects, 0.05 eV level of error for band gap.
3. EH and EH2 : For si, if EH and EH2 are the same, the following error occurred.

* fexit,fexit2,fexit3 error retval= -1  
   Exit -1 zhevUUUUUUtk2: nev /=nevx something wrong.
* The large EH, EH2 get to be meaningless. We usually use up to . (If you use very large EH such as E, I am not so sure weather it is )

1. The options about the product basis within MT. (SeungWoo’s memo)

* <PRODUCTUUUUUUBASIS>  
   tolerance to remove products due to poor linear-independency  
   0.100000D-02 ! =tolopt
* When the product basis are made, we may have poorly linear independent basis. For example, one in the set {} would be almost give by a linear-combination of others. We need to make the linear-independent set. Therefore, after calculating the overlap matrix . We do diagonalization, then we remove eigenvectors corresponding to small eigenvalues than tolopt. See the **hbasfp0** command in **gwsc**
* lcutmx(atom) = maximum l-cutoff for the product basis.  
   4 4 4 2 2 4 4
* For $\phiUUUUUU1 \times \phiUUUUUU2$ case, . So ‘lcutmx’ changes the maximum cutoff for the l. The order is the same as the order of atoms in the **ctrl** file.
* atom l nnvv nnc !  
   1 0 3 3  
   1 1 3 2  
   1 2 2 1
* ‘atom’ means the atom number identified in the **ctrl** file.  
  ‘l’ is the angular momentum quantum number.  
  ‘nnvv’ is the number of radial functions (valence) on the augmentation-waves.  
  ‘nnc’ is the number of radial functions for core.  
  The latter two ones, ‘nnvv’ and ‘nnc’, will be understood more clearly if you see the following ones.
* atom l n occ unocc ! Valence(1=yes,0=no)  
   1 0 1 1 1 ! 5SUUUUUUp -----  
   1 0 2 0 0 ! 5SUUUUUUd  
   1 0 3 1 1 ! 4SUUUUUUl  
   1 1 1 1 1 ! 5pUUUUUUp  
   1 1 2 0 0 ! 5pUUUUUUd  
   1 1 3 1 1 ! 4pUUUUUUl
* Above options are about the product basis set within MT (Valence).  
  ‘atom’ and ‘l’ are explained above. ‘nnvv’ for ‘atom = 1 and l = 0’ was ‘3’ so this case we have 3 basis (‘n = 1, 2, 3’)  
  ‘n’ is the degree of freedom of the radial function, . ‘n = 1’ means , ‘n = 2’ means , and ‘n = 3’ means kind of , which the dot above the letter represents the differentiation with respect to the energy. So ‘n = 1 and 2’ is related to the linearization of the radial function and ‘n = 3’ is the local orbital which is restricted in MT. The local orbital can be modified changing ‘PZ’ in the **ctrl** file. Finally, the number of the basis set which is needed for expanding eigenfunctions is .  
  ‘occ’ and ‘unocc‘ mean that we use only ones that checked as ‘1’, in other words we neglects ‘0’ cases for making product basis. Be careful for confusion with name ‘occ’ and ‘unocc’. These don’t mean that occupied or unocc. When making product basis, $M = \phiUUUUUU1 \times \phiUUUUUU2$, ‘occ’ corresponds to $\phiUUUUUU1$ and ‘unocc’ to $\phiUUUUUU2$. For example,
* atom l n occ unocc ! Valence(1=yes,0=no)  
   1 0 1 1 1 ! 5SUUUUUUp -----  
   2 3 1 0 1 ! 4fUUUUUUp
* If the options are like the above, the product basis will be consists of ($\phiUUUUUU1 = \phiUUUUUU{atom=1,l=0}) \times (\phiUUUUUU2 = \phiUUUUUU{atom=1,l=0})$, ($\phiUUUUUU{atom=1,l=0} \times \phiUUUUUU{atom=2,l=3}$). As you can see, ($\phiUUUUUU1 = \phiUUUUUU{atom=2,l=3}$) is skipped.  
  In the **ctrl** file, ‘EH’ controls the l part. As for ‘EH’, (s, p, d, f) are used but **GWinput** file uses (s, p, d, f, g). ‘EH’ : HEAD part. ‘GWinput’ : contains TAIL part... need more explanation.
* atom l n occ unocc ForX0 ForSxc ! Core (1=yes, 0=no)  
   1 0 1 0 0 0 0 ! 1S -----  
   1 0 2 0 0 0 0 ! 2S  
   1 0 3 0 0 0 0 ! 3S
* Above options are about the product basis set within MT (Core).  
  ‘nnc’ for ‘atom = 1 and l = 0’ was ‘3’ so this case we have 3 basis (‘n = 1, 2, 3’)  
  Finally, for the convergence check, we can modify the following three things, (i) tolerance, (ii) lcutmx, and (iii) occ and unoccu.

== one show QSGW (not one-shot GW) ==  
 one-shot QSGW can be useful in cases.  
 As it contains off-diagonal part, we can resolve band tanglement  
 problem in Ge (no band gap).  
  
== Restart calculation in lda ==  
 lmf(lmf-MPIK) read rst.\* in defaluts.  
 rst contains electron density.  
 If rst is already converged, it stops after two iteration.  
 rst contains atomic positions.  
 So, in order to read atomic positions change in ctrl,  
 Use options shown in lmf --help.  
  
== Restart calculation in qsgw ==  
 To remove mixsigm\* (mixing for sigm), maybe required.  
  
== iteration check ===  
 First, watch console output of gwsc (do redirect to output file)  
 Need to check OK! signs arrayed on 1st columns.  
  
 gwsc iteration is time cosuming,  
 So we need to check calculations are normally going on or not.  
  
 Memory inefficiency.  
 Set 'KeepEigen off' an 'KeepPPOVL' off.  
 In fact, out code is still inefficient for memory usage.  
   
 grep gap llmf ---> minimum gap at mesh point.  
 see save.\* ,or grep '[xc] ' save.\*  
 the end of iteration of lmf is shown as x or c.  
 (if failed, QPU file.   
 dqpu QPU.4run QPU.3run  
 As for usual semi-conductor, accuracy abou t0.1 eV is limit of current implementation.  
 Set vwn (xcfun=1) looks better (stable) for GW.  
  
 $grep rms lqpe\*   
 shows  
 ... rmsdel=2.44D-04  
 ... rmsdel=4.91D-03  
 ... rmsdel=2.44D-04  
 ... rmsdel=3.37D-04  
 If rsmdel is getteing to be smaller, it is on convergence path.  
 (but in magnetic cases, it may give be too good even not yet going to  
 be converged..., beccause magnetic energy is so small)  
  
 grep diffe llmf ---> difference of energies of each iteration.  
  
 ehf (harris energy)  
 ehk (Hohenberg kohn energy)  
  
== emax cutoff for APWs. ==  
 We can not use so many APWs in current version,  
 because of overcompleteness (this is because null vector within MTs),   
 In anyway, use pwemax=3 as standard (test it with 4 or 5).  
 To avoid failure of calculation, we may use smaller MT radius for  
 alkali, and alkali-earth elements.   
 In feature, I think we can introduce pseudopotentials for these atoms only.  
  
== Check Used MTO   
 Near beginig of console output, what MTO you use is shown as: (GaAs case).  
 sugcut: make orbital-dependent reciprocal vector cutoffs for tol= 1.00E-06  
 spec l rsm eh gmax last term cutoff  
 Ga 0\* 1.13 -1.00 6.579 1.19E-06 1459  
 Ga 1\* 1.13 -1.00 7.028 1.26E-06 1807  
 Ga 2\* 1.13 -1.00 7.475 1.09E-06 2109  
 Ga 3 1.13 -1.00 7.920 1.06E-06 2637  
 Ga 0\* 1.13 -2.00 6.579 1.19E-06 1459  
 Ga 1\* 1.13 -2.00 7.028 1.26E-06 1807  
 Ga 2 1.13 -2.00 7.475 1.09E-06 2109  
 As 0\* 1.18 -1.00 6.300 2.13E-06 1243  
 As 1\* 1.18 -1.00 6.720 1.26E-06 1471  
 As 2\* 1.18 -1.00 7.140 1.37E-06 1837  
 As 3 1.18 -1.00 7.558 1.05E-06 2229  
 As 0\* 1.18 -2.00 6.300 2.13E-06 1243  
 As 1\* 1.18 -2.00 6.720 1.26E-06 1471  
 As 2 1.18 -2.00 7.140 1.37E-06 1837  
  
== gwsc cause error stop.  
 Have you ever changed MTO setting? Consistent with GWinput?  
  
== QSGW for Fe.  
 It is better to use 3p as core. Furthermore, 3d+4d as valence is better.   
 Thus we need to set PZ=0,3.9,4.5  
 I also got aware that emaxUUUUUUsigm should be large enough (4$\sim$5 Ry)   
 to have smooth band dispersion. n1n2n3 can be 10x10x10.  
  
== RSRNGE: enlarge RSRNGE ===  
 Use RSRNGE=10 or so (in cases, RARNGE=20 or more is required),   
 for large number of k points. Try and enlarge it if it fails with a  
 message "Exit -1 rdsigm: Bloch sum deviates more than allowed tolerance (tol=5e-6)".  
 We will have to make it automatic in future.  
 Detailed memo (for deverlopers) is at the bottoem of ecalj/Document/BACKUP/MarksOriginalDoc/gw.html.  
  
== Q0P check  
 In cases, it is better to use Q0Pchoice=2 instead of default Q0Pchoice=1.  
 (For slabs, Q0Pchoice=2 may be better; need check more. In anyway,  
 it is problematic to use unbalanced k points for anisotropic cell).  
 See Copmuter Physics Comm. 176(2007)1-13).  
  
=== When calculation in LDA level fails ===  
when calculation fails in LDA level.  
 (1) smaller MT  
 (2) fewer PW. smaller pwemax.  
 (3) core as semicore.  
  
  
=== LDA+U ===  
not yet written...  
  
=== MAE by rotating crystal ===  
(we have a sample at lm7K/TESTsmaples/MAEtest/, but only in GGA/LDA).  
  
=== spin wave ===  
J calculation.  
  
  
====  
If not stable convergence in gwsc, try to set  
mixbeta 0.5  
(and/or mixpriorit 3 or something)  
at the begining of sigma.  
  
  
=======  
cleargw (directory):  
This command clean up up intermediate files under (directory).  
This recursively into deeper level. Be careful, or edit it.  
I use it as '>cleargw .'  
  
  
  
------------------------------------  
Magnetic moment within MTs are shown as  
------------------------------------  
 charges: old new   
 smooth 17.240314 17.240740 ...  
 mmom 0.000024 -0.000010   
 site 1 6.207135 6.206590   
 mmom 1.062276 1.062991 <--- here  
 site 2 6.207115 6.206834   
 mmom -1.062323 -1.062958 <--- here  
 site 3 1.172718 1.172918   
 mmom 0.000011 -0.000011   
 site 4 1.172718 1.172918   
 mmom 0.000011 -0.000011   
In this case, MTsite1 has 1.062991 and MTsite2 has -1.062958.  
>grep 'lin mix' -A30 llmf   
can take out this message (if console output is in llmf).  
  
  
-----------------------------  
ORBITAL MOMENT in pertubation:  
-----------------------------  
Try   
>lmf nio --rs=1,0 -vso=1 --quit=band >llmf  
After converged, try  
>grep IORBTM -A20 llmf  
Then llmf shows shows orbital moment in first order perturbation.  
(Here --rs=1,0 read rst.\* file but not change it. See >lmf --help.  
--quit=band means quit just after band calculation.)  
  
== EPS mode,  
 Check Im part of chi0 is smoothly damping at high energy (typically  
 1Ry or larger enengy range). If there is some large Im part remains,  
 something strange (usually due to orthogonality problem of  
 eigenfunctions when you set low q).  
  
 Related source codes are in ecalj/lm7K/ .  
 A command ecalj/lm7K/ctrlgenM1.py can generate 'standard input file (ctrl file)'   
 just from a given crystal structure file called as ctrls file.   
 Binaries are lmf and lmf-MPIK (MPI k-parelell verion).

## lmf –help

lmf –help show option of –rs=(five numbers); this let lmf know how to read atm.\* file which is the initial atom file by lmfa.

# Wannier function

We can generate Wannier functions (maximally localized Wannier Functions or similar) by a script genMLWF. It automatically perform cRPA calculation sucessively. (If it is not necessary, insert ’exit’ in genMLWF, after it performs 2@).

Try to run examples in /MATERIALS/SampleUUUUUUMLWF/@. Read in it. To run the script genMLWF, we need to get by editing .tmp@. (2@ contains default Wannier section). In addition, we have some settings (energy windows and so on). This is the example of the initial conditions for Cu case. 5 is the number of Wannier function. The most left one means index and the right one of it is index. They are written in the **@MNLA** file.

Then we can run genMLWF. After it finished, we can analyze it results. (if you don’t need Wannier funciton plot, You can skip a line of wanplot in genMLWF. Then we don’t need to set visUUUUUUwanUUUUUU\* options.)

## lwmatK1 and lwmatK2

If you input the following command

>grep Wan lwmatK\*

You will get the following results. (This case : Cu cases)

lwmatK1: Wannier 1 1 24.644475 0.000000 eV  
lwmatK1: Wannier 1 2 24.644576 0.000000 eV  
lwmatK1: Wannier 1 3 25.471361 0.000000 eV  
lwmatK1: Wannier 1 4 24.644575 0.000000 eV  
lwmatK1: Wannier 1 5 25.470946 0.000000 eV  
lwmatK2: Wannier 1 1 0.000000 eV -21.263759 -0.000000 eV  
lwmatK2: Wannier 1 2 0.000000 eV -21.263839 0.000000 eV  
lwmatK2: Wannier 1 3 0.000000 eV -21.931033 -0.000000 eV  
lwmatK2: Wannier 1 4 0.000000 eV -21.263839 -0.000000 eV  
lwmatK2: Wannier 1 5 0.000000 eV -21.930702 -0.000000 eV

### Wanneir Branch now under developing (imported from T.Miyake's Wannier and H.Kino's).  
 A. make at ecalj/fpgw/Wannier/ directory, and do make, and make install.   
 (need to check Makefile first). You first have to install fpgw/exec/ in advance.  
 B. Samples are at these directories.   
 MATERIALS/CuMLWFs (small samples),  
 MATERIALS/CuMLWF/  
 MATERIALS/CuMLWFs/  
 MATERIALS/FeMLWF/   
 MATERIALS/NiOMLWF/  
 MATERIALS/SrVO3MLWF/  
 C. With GWinput and ctrl.\*, run   
 >genMLWF  
 at these directories.  
 In GWinput, we supply settings to generate Wannier funcitons. (Sorry,not documentet yet..)  
 D. After genMLWF, do  
 >grep Wan lwmatK\*  
 then compare these with Result.grepWanlwmatK  
 These are onsite effective interactions (diagonal part only shown).  
 \*.xsf are for plotting the Maximally localized Wannier funcitons.  
Anyway, documentaion on Wannier is on the way half.  
Time consuming part (and also the advantage) is for effective interaction in RPA.  
Look into the shell script genMLWF; you can skip last part if you don't need the effective interaction.

# ctrl file details

A ctrl file is usually generated from a ctrls file by the ctrlgenM1.py (a crystal structure file is not “ctrl” but “ctrls”.). It contains self explanation. Here we give complementary explanations to it. Let us Look into a ctrl file. This is a head part of ctrl.cu generated by ctrlgenM1.py:

### This is generated by ctrlgenM1.py from ctrls   
 ### For tokens, See Document/Manual/CategoryAndToken.org.  
 ### Do lmf --input to see all effective category and token ###  
 ### It will be not so difficult to edit ctrlge.py for your purpose ###  
 VERS LM=7 FP=7 # version check. Fixed.  
 IO SHOW=T VERBOS=35 TIM=2,2  
 # SHOW=T shows readin data (and default setting at the begining of   
 console output)  
 # It is useful to check ctrl is read in correctly or not  
 (equivalent with --show option).  
 # larger VERBOSE gives more detailed console output.  
 SYMGRP find # 'find' evaluate space-group symmetry automatically.  
 # Usually 'find is OK', but lmf may use lower symmetry  
 ...

Note that # means comment lines. We can also use lines starting from % const ... to define variables and set constant.

We see “categories” such as VERS, IO, and so on. The beginning of categories are starting from the first column. Under categories, we have "tokens" such as VERBOSE. Thus we specify full name of token VERBOSE under category IO as IOUUUUUUVERBOSE.

* IOUUUUUUTIM is for debugging. It shows which subroutines are called and so on. Bigger number shows deeper subroutines.
* SYMGRP is a category without token under it; we set generators of space group (See explanation in previous paragraph). When we set find, it automatically calculate symmetry of crystal lattice. If we like to enforce symmetry, set some of generators which are shown by lmchk.
* We see ctrls is embedded in the ctrl by ctrlgenM1.py.
* ... (skip) ...  
   % const da=0 alat=6.798  
   STRUC ALAT={alat} DALAT={da}  
   PLAT= 0.0 0.5 0.5 0.5 0.0 0.5 0.5 0.5 0.0  
   NL=4 NBAS= 1 NSPEC=1  
   SITE ATOM=Cu POS=0 0 0  
   ... (skip) ...
* NL, NBAS(number of SITE) and NSPEC(number of SPEC) are automatically added by ctrlgenM1.py. It is possible to deform unit cell by adding some optional tokens under STRUC category. Search STRUC in an old document (which may be still effective) BACKUP/MarksOriginalDoc/lmto.html. However, it is a little complicated. For new calculations, it is better to find some examples first.
* **SITE** category: As for MT sites, we have two categories. (1)SPEC(species) and (2)SITE(specify centers of atoms(species) in primitive cell). As for SPEC, we specify MTs(radius, Z, MTOs on it) appeared in the cell. These are defined subtokens under SPEC=foobar (we have multiple SPEC=foobar).
* Then we place these MTs at SITE sections in the cell. At SITE, we specify atomic sites (What SPECis placed to positions by POS) in a primitive cell. We set POS= by direct form (Cartesian) but with the unit of ALAT+DLAT. Total number of SITE (number of tokens SITE) is the number of atoms in the primitive cell. Setting POS= under SITE=foobar means that we place MT named as foobar defined in SPEC=foobar. In addition, we can set SITE, if you like to find relaxed structure (we simultaneously set DYN category) in LDA. As for relaxation, see LaGaOUUUUUUrelax/ctrl.lagao example, and read DYN in Document/Manual/CategoryAndToken.org.
* The SITE=foobar (with same foobar with different POS) are not necessarily equivalent with respect to the space group operation of a system. Thus SITEUUUUUUATOM=foobar are divided into “classes” which are connected by the operation. The lmf automatically judge “classes” (see also info by lmchk). Thus not need to specify it, but it may be better to check it. A sample is lmchk lagao at ~/ecalj/lm7K/TESTsamples/LaGaOUUUUUUrelax
* **SPEC** category: In ctrls, we have not yet specified contents of SPEC; we have just given default symbols or only Z= when we use non-default names (shown by ctrlgenM1.py –showatomlist). The command ctrlgenM1.py adds default SPEC sections.
* We have some SPECUUUUUUATOM, under which we give subtokens such as SPECUUUUUUATOMUUUUUUR(MT radius), SPECUUUUUUATOMUUUUUUZ(nucleus charge), cutoff parameters of angular momentum, and so on. These SPECUUUUUUATOM is refereed to in SITE.
* An example of SPEC category is
* SPEC   
   ATOM=Fe Z=26 R=1.70   
   KMXA={kmxa} LMX=3 LMXA=4 NMCORE=1   
   PZ=0,3.9,4.5  
   EH=-1 -1 -1 -1 RSMH=0.85 0.85 0.85 0.85   
   EH2=-2 -2 -2 RSMH2=0.85 0.85 0.85  
   MMOM=0 0 2 0   
     
   ATOM=... (then the similar block of ATOM= are repeated.)  
   ...
* Under the token ATOM=Fe, we have subtokens SPECUUUUUUATOMUUUUUUZ,SPECUUUUUUATOMUUUUUUR, and so on.
* Subtokens Z= is the nucleus charge and R= MT radius. Note that Fe is just a name to distinguish MT sphere in the cell. If you set SPEC=27, it is recognized as Co (since Z=27). LMX=3 is the maximum l of MTOs. Thus maximum l of MTO is l=3. The maximum of l to expand electron density and potential within MT is LMXA (in contrast to usual LAPW), we can use quite small LMXA such as LMXA=4. NMCORE=1 means we calculate core density without non magnetically-polarization. This can reduce computational confusion.  
  PZ is to set local orbital (if not, no local orbitals). EH and RSMH are to specify first set of MTOs.(We can check how local orbitals are set by lmfa explained in the next section). EH2 and RSMH2 are to specify second set of MTOs.  
  After PZ=, we have three numbers. These are numbers for s,p,d,f,g,... channels. Zero means not exist. You can use space or comma(,) as delimiter. Here not only the integer part of principle quantum number, but also the fractional part should be supplied (If PZ=0,3,4, it does not work.) Now PZ=3.9 for p and PZ=4.5 for d. This means we use local orbital for 3p, and local orbital for 4d (fractional parts (continuous principle quantum number) are large for core like orbital, and smaller for extended orbital or something. See Logarithmic Derivative Parameters in Document/BACKUP/MarksOriginalDoc/lmto.html. This is a little confusing, thus we will explain this in appendix. See Sec.[10.1](#pzsetting).  
  EH(damping factor), and RSMH (where the smooth Hankel function bent) determines MTOs (or its envelope function as a smooth Hankel function). We now set four numbers for them. Thus we set MTOs s,p,d,f with EH=-1 and RSMH=0.85. Our current test shows that RSMH is one half of R (that is, 0.85=1.70/2, but minimum RSMH is 0.5) and not need to be dependent on s,p,d,f. (If LMX=2, s,p,d are allowed and no f MTOs.) EH is -1; not need to change except test purpose. In a similar manner, EH2 and RSMH2 for second set of MTOs are given. Just three numbers means these for s,p,d.  
  MMOM=s,p,d,f... gives initial condition of magnetic moment in $\muUUUUUUB$ (number of up-down electron).  
  In cases such as As, the local orbital given by default ctrl is responsible of rather deep core, and it is not need to be treated as valence electrons. In such a case, we don’t need local orbital.  
  In the case of AntiFerro-II NiO, it contains two NiO in a primitive cell. Thus it is reasonable to have two SPEC as Ni1 and Ni2, although subtokens under ATOM=Ni1 and ATOM=Ni2 (e.g. SPECUUUUUUATOMUUUUUUEH for them) are the same except initial condition of magnetic moment of MMOM=s,p,d,f... See example of NiO.

The minimum help of call Categoryare listed with minimum explanation with

$ lmf --input

It gives a long output. But many of them are experimental and not need to manage them. A part of it is

Token Input cast (size,min)  
 ------------------------------------------  
 ... ...  
  
 STRUCUUUUUUALAT reqd r8 1, 1  
 Scaling of lattice vectors, in a.u.  
 ... ..

This is an minimum explanation of it. "reqd" means "required" (no default). r8 means it read with real number, 1,1 means that ALAT=xxx should contain one number minimum (max is also one) (See also STRUC, and so on).

There are kinds of examples in ecalj packages. Please look into their ctrls.\* and ctrl.\* These are in lm7K/TESTsample/\* and ecalj/CMDsampls. In addition, ecalj/MATERIALS contain many samples (need a command); see a later subsection.

As for what is shown in $ lmf --input, most of important tokens are already described in the ctrl file generated by ctrlgenM1.py. So, we don’t need to care many options shown by it.

For QSGW calculation:

We need a setting in ctrl file to read sigm file (HAM). It is simplified now, and not need to care it so much. As we set RDSIG=12 in defaults, lmf read sigm file and add it to one-body potential as long as sigm.\* exist.  
**NOTE for old users**: We now set SIGP[MODE=3 EMAX=9999.] in ctrl file to read self-energy in lmf (or lmf-MPIK). This is because we use very localized MTOs (similar with the Maxloc Wannier). Our test shows reasonable results and this simplify algorithms. In my previous version, we asked you to use SIGP[MODE=3 EMAX=2.0] where EMAX is a little (0.5Ry) less than emaxUUUUUUsigm. If something strange occurs, try this setting).

In principle, QSGW result should not depended on the choice of XCFUN. However, it can affect slightly. In our tests, it seems slightly better to use VWN (XCFUN=1) for QSGW calculations. (BUT need to check more...)

## How to set local orbitals

As we stated, do "lmfa |grep conf" to check used MTO basis.   
  
 We have to set SPECUUUUUUATOMUUUUUUPZ=?,?,?   
 (they ordered as PZ=s,p,d,f,... ) to set local orbitals.  
   
 lmv7 (originally due to ASA in Stuttgart) uses a special terminology  
 "continious principle quantum number for each l", which is just  
 relatated to the logalismic derivative of radial funcitons at MT  
 boundary. It is defined as  
 P= principleQuantumNumber + 0.5-1/pi\*atan(r\* 1/phi dphi/dr),  
 where phi is the radial function for each l. For example,   
 P= n.5 for l=0 of free electron (flat potential) because phi=r^0,  
 P= n.25 for l=1 because phi=r^1;   
 P= n.147584 for l=2 because phi=r^2; P=, n.102416 n.077979 for l=3,4.  
 (Integer part can be changed). See Logarithmic Derivative Parameters in  
 http://titus.phy.qub.ac.uk/packages/LMTO/lmto.html#section2  
  
 Its fractioanl part 0.5-atan(1/phi dphi/dr) is closer to unity for  
 core like orbital, but closer to zero for extended orbitals.  
  
 Examples of choice:  
 Ga p: in this case, choice 0 or choice 2 is recommended.  
 We usually use lo for semi-core, or virtually unoccupied level.  
  
 (0)no lo (4p as valence is default treatment without lo.)  
 3p core, 4p valence, no lo: default.  
 Then we have choice that lo is set to be for 3p,4p,5p.  
 (1)3p lo ---> 4p val (when 3p is treated as valence)  
 3d semi core, 4d valence   
 Set PZ=0,3.9   
 (P is not requied to set. \*.9 for core like state. It is just an initial condition.)  
 (2)5p lo ---> 4p val (PZ>P)  
 Set PZ=0,5.5   
 5.5 is just simply given by a guess (no method have yet  
 implemented for   
 If 5.2 or something, it may fails  
 because of poorness in linear-dependency. We may need to observe  
 results should not change so much on the value of PZ.  
  
 (3xxx)4p lo ---> 5p val (we don't use this usually. this is for test purpose)  
 4p lo, 5p valence   
 Set PZ=0,4.5 P=0,5.5 (In this case, set P= simultaneously).  
 (NOTE: zero for s channel is to use defalut numbers for s)  
  
 Ga d: (in this case, choice 0 or choice 1 is recommended).  
 (0)no lo (3d core, 4d valecne, no lo: default.)  
 Then we have choice that lo is set to be for 3d,4d,5d.  
 (1) 3d lo ---> 4d val (when 3d is treated as valence)  
 Set PZ=0,0,3.9 (P is not required to set)  
 (2) 5d lo ---> 4d val (PZ>P)  
 Set PZ=0,0,5.5  
 (3xxx) 4d lo ---> 5d val (this is for test purpose)  
 Set PZ=0,0,5.5 P=0,0,4.5  
 (NOTE: zero for s,p are to use defalut numbers )  
  
 If you like to read from atm.ga file instead of rst file(if exist).  
 You have to do lmf --rs=1,1,0,0,1, for example. See lmf --help  
 Becase rst file keeps the setting of MTO, thus change in ctrl is not  
 reflected without the above option to lmf.  
=============================================================

# GWinput details

## **generate a template of GWinput**

As in the previous section, we need two input files ctrl.si and GWinput. In principle, these two determines final results uniquely. A template GWinput.tmp is generated by mkGWIN2. Required files are

: The control file for PMT method.  
(Recently modified mkGWIN2 runs lmfa internally. If you use older version, do lmfa in advance).

: A file including computational conditions for the calculation. In addition, it specifies the **k** points for which you calculate the QP energy.

When you run **mkGWIN**, it asks you to supply three numbers for BZ integration as

== Type three integers n1 n2 n3 for Brillouin Zone meshing for GW! ==  
 n1=

Then you need to type a number e.g. as "2 " for n1. Then you need to repeat it for n2 and n3 as  
n1= 2  
n2= 2  
n3= 2  
. These numbers specifies what k points in BZ is used for BZ integration (In this case, **k** point in the 1st BZ is used. Roughly speaking, we need (or ) to get band gap for Si and so on, with eV accuracy.)

Then you have to edit GWinput.tmp and copy it to GWinput. We details the GWinput in later chapter.

We need to repeat mkGWIN2 when you change MTO sections in ctrl file (adding PZ case, and so on).

## overview of GWinput

(Because of historical reason, input file is different from ctrl.\*).

The main input files is GWinput. This controls the setting of calculation. The file GWinput consists of structures as  
*keyword1 data1*  
*keyword2 data2*  
...  
In each lines, it consists of keyword and data. Data can be single or plural. As for keywords, upper case or Lowercase is not distinguished. All keywords should start from 1st column (no space at head). Order of lines are irrelevant. As for logical variable, you can use anything “true, yes, on, 1, T” for .true., and anything “false, no, off, 0, F” for .false.  
Or we have “tag sections” in GWinput specified by <PRODUCTUUUUUUBASIS>@, <QPNT>@, <PBASMAX>@, <QforEPS>@, and <QforEPSL>@. (<PRODUCTUUUUUUBASIS>@ is requires for all kinds of calculations. <PBASMAX>@ is optional. <QforEPS>@ and/or <QforEPSL>@ are required for epsilon mode). It is like

<PRODUCTUUUUUUBASIS>  
 tolerance to remove products  
 0.100000D-02 ! =tolopt  
 lcutmx(atom)   
 3 3   
 atom l  
...  
</PRODUCTUUUUUUBASIS>

. In these tag sections, you have to keep format for its own (usually numbers are read by free format (\*,\*)@).

The fundamental readin routine for GWinput is a subroutine defined in /keyvalue.F@. is a general and convenient readin routine in full use of the f90 features. Read a head part of the file and try to do "grep getkeyvalue \*.F" in /@ or /@ so as to see how to use it (test routine is /kinoUUUUUUinputUUUUUUtest.F@.)

Soche the GWinput consists of three sections

1. General section
2. <PRODUCTUUUUUUBASIS>@ section
3. <QforEPS>@,<QforEPSL>@ section (only effective for dielectric function mode)
4. <QPNT>@ section (only effective for one-shot mode)
5. <PBASMAX>@ section (optional)

We will explain each by each in the followings.

## General section

In general section, it looks like

! ##### From GWIN0 ################   
n1n2n3 1 1 1 ! for BZ meshing in GW   
QpGcutUUUUUUpsi 4.000 !(See unitUUUUUU2pioa for unit) |q+G| cutoff for eigenfunction.  
QpGcutUUUUUUcou 3.000 !(See unitUUUUUU2pioa for unit) |q+G| cutoff for Coulomb and W.  
unitUUUUUU2pioa off ! off --> a.u.; on--> unit of QpGcutUUUUUU\* are in 2\*pi/alat   
alphaUUUUUUOffG 1.000 !(a.u.) Used in auxially function in the offset-Gamma method.  
!emaxUUUUUUchi0 99999.000 !(Ry) emax cutoff for chi0 (Optional)  
emaxUUUUUUsigm 3.000 !(Ry) emax cutoff for Sigma  
  
! ##### FREQUENCIES from GWINUUUUUUV2 ################   
dw 0.005000 !(a.u.) energy-mesh (bin width size) along real axis.  
omgUUUUUUc 0.040 !(a.u.) energy-mesh is twiced at omgUUUUUUc  
 ! coaser mesh for higher energy. Width get to be doubled at omgUUUUUUc.  
iSigMode 3 ! QSGW mode switch for gwsc. use =3.  
niw 10 ! Number of frequencies along Im axis. Used for integration to get SigmaUUUUUUc  
 ! E.g. try niw=6 and niw=12  
delta -0.10D-05 !(a.u.) Broadening of x0. negative means tetrahedron method.  
 ! used by hx0fp0. You get smeard x0 witth abs(delta).  
deltaw 0.020000 !(a.u.) Mesh for numerical derivative to get the Z factor  
esmr 0.003000 !(Ry) used by hsfp0. Keep esmr smaller than band gap for insulators  
 ! Poles of G^LDA are treated as if they have width esmr in hsfp0.   
 ! Change esmr for metals. See DOSACC\*---especailly around Ef.  
GaussSmear on ! Gaussian or Rectangular smearing for Pole of G^LDA with esmr for hsfp0.

—————————————————

1. BZ integration.  
   3 integers as (no default); They are .  
   Brillouin Zone mesh for integration is determined by keywordsBZmesh and n1n2n3. Current version only allow regular mesh point including Gamma point for . But not that Chibelow allow you to use off-Gamma mesh for (and dielectric function mode).

* We usually take ’4 4 4’, ’6 6 6’ or ’8 8 8’ for GaAs. For metal such as Fe, ’10 10 10’ or more is better.
* (on or off)
* Chi= on (default): Use regular mesh (including gamma) for eps calculation.
* Chi= off : Use off-Gamma mesh (Not including gamma) for eps calculation.
* (In cases, Chioff gives faster convergence as for n1n2n3; not only for GW, but also for dielectric functions eps.)

1. Plane wave (${\bf q+G}$) cutoff  
   1 real (no default)  
   1 real (no default)  
   1 logical (no defalt)  
   We have two cutoff for ${\bf q+G}$. QpGcut is the cutoff of for the IPW in the expansion of the eigenfunctions. QpGcut is for the IPW of the interactions . Its unit is specified by unit2pioa ; "off" means unit in a.u. and "on" means in unit of $\frac{2 \pi}{\mbox{\rm alat}}$. (alat is length scale unit in ctrl.\*).  
   Rule of thumb: QGcutUUUUUUpsi is a little (usually 0.5 or so) larger than QpGcutUUUUUUcou. It becomes accurate if we use large QpGcutUUUUUUcou. But it enlarge size of IPW(interstitial plane wave) part of Mixed product basis. For test, try 2.7, 3.2, 3.7 for QGcutUUUUUUcou (and add 0.5 or 1 for QGcutUUUUUUpsi). Larger one is expensive.  
   We expand eigenfunctions in the Muffin-tin division of the space. See Eq. in Ref.; in the current implementation, we use very simple form of eigenfuncitons (not by the 3-component formalism in the ).

* Thus the form of expansion is just related to the division of space; not directly related to the difference among LAPW, LMTO, and PMT.

1. Eigenfunctions within MTs (no parameters setting for them).  
   The radial functions (phi and phidot for each ), corresponding to the true parts, (= 2nd component in the 3-component formalism ), are automatically determined already in the one-body part of program lmf-MPIK.
2. Cutoff for used bands.  
   : 1 real (optional,default=), in Ry  
   : 1 real (optional,default=; We usually use 3 Ry).  
   emaxUUUUUUsigm is the maximum limit of the self-energy (measured from the Fermi energy). See the paper which shows how the results are affected by emax. But in cases, small emaxUUUUUUsigm can give poor dispersion curve (slightly unnatural behavior) because of sudden cutoff by emax. However, we like to use smaller value to reduce computational time.

* That is, larger is better, but expensive (And note that we simultaneously need to use empty spheres when we use large emax, as shown in , ).
* Generally speaking, accuracy less than 0.1eV (for bandgap) is allowance of current technique. Probably, it may be possible to have better accuracy, but it may ask us to repeat many calculations with changing conditions to confirm stability.
* : 1 integer (optional,default=)  
  : 1 integer (optional,default=)  
  These specify how many bands you use in **hx0fp0** (for chi0) and in **hsfp0** (for sigma). Higher bands above them are neglected.

1. Energy mesh related parameters.  
   — new after march2016 —  
   : 1 real (a.u.). Mesh width along real axis for .  
   : 1 real (a.u.).  
   HistBinUUUUUUdw and HistBinUUUUUUratio specify real space bins which we accumulate imaginary part weight of polarization functions. The bins are (see frhis in ecalj/fpgw/gwsc/mUUUUUUfreq.F)

$$\begin{aligned}
\omegaUUUUUUi=b\*(\exp(a\*(i-1))-1),\end{aligned}$$

1. where $[\omegaUUUUUUi, \omegaUUUUUU(i+1)]$ (+1@) is the -th bin. HistBinUUUUUUdw is bin width at . The ratio $\omegaUUUUUU(i+1)/\omegaUUUUUU(i)$ for large is =HistBinUUUUUUratio. This choice of getting coarser at high energy is because we think around gives most important contribution to the GW approximation. If histogram bins are too wide, dielectric function can be less accurate, but results may be not so much affected.

* In GW calculation, the Plasmon pole is important. It is determined not by the Drude weight; it usually gives small contribution to the Plasmon pole (for example, Si is described well by a Plasmon pole model, but Si has no Fermi surface). We expect that the GW results are not so sensitive to the choice of HistBinUUUUUUdw, HistBinUUUUUUratio usually. We may use fine mesh when we plot quantities such as near .
* The ecalj gives for metal; where is the effective interaction averaged in the -cell . And get closer to for larger .
* Default value of them are given in fpgw/gwsrc/mUUUUUUfreq.F. They are
* call getkeyvalue("GWinput","HistBinUUUUUUratio",oratio, default=1.03d0)   
   call getkeyvalue("GWinput","HistBinUUUUUUdw",dw, default=1d-5) !a.u.
* These are safer settings. Thus new mkGWINUUUUUUlmf2 (GWinput.tmp generator) gives default lines in GWinput as
* HistBinUUUUUUdw 2d-3 ! 1d-5 for metal !(a.u.) bin width along real axis at omega=0.  
  HistBinUUUUUUratio 1.08 ! 1.03 for check. frhis(iw)= b\*(exp(a\*(iw-1))-1),  
   where a=ratio-1.0 and dw=b\*a
* . This is given in the conv2gwinput.F.
* — old version. before 14march2016 –  
  : 1 real (a.u.). Mesh width along real axis for .  
  : 1 real (a.u.).  
  dw and omg determines mesh along real axis to calculate ...  
  ————————-  
    
  : 1 real (a.u.). We usually use very small number as -1d-8 for gw, eps mode and so.  
  xxx does this really make the stabilization? xxx This is the size of in denominator of (EQ.xxx). But (I think) we can (or can not) use it so as to make broadening for theoretical test (maybe not exactly corresponding to ). or when you make calculation stabilized (Takao need to check this point again xxx!)  
  [Old note. Need check xxx: In gw, it is used for broadening of x0 when it call hx0fp0. Then delta is is EQ.32. The sign of delta is just used as a flag whether you use the tetrahedron method of dielectric constant or not; minus sign means "Use the tetrahedron method for "; plus sign means you do it by simple sum. You can usually use this default setting. But it might be possible to use a larger value to smear the fine structures on the energy-dependence of in cases. This might be necessary if is so energy-dependent and is not so small to resolve the structure —but I don’t know.]  
  : 1 integer.  
  Number of integration points along the imaginary axis(FIG.1) to get $\SigmaUUUUUUc$. See routines wint\* called from sxcf\*.F, which is called from the main routine hsfp0.m.F (or hsfp0.sc.m.F in the QSGW case). The integration points are , where is the usual Gaussian-integration points for the interval [0,1]. In addition, we give the special analytical treatment for the peaky part at . Out tests shows niw=6 for Si is good enough for 0.01 eV accuracy. The convergence as for niw is quite good. This integration scheme has been developed by Ferdi Aryasetiawan. The number of points should be the one of 6,10,12,16,20,24,32,40,or 48. It is because we use a subroutine gauss in /gwsrc/mate.F prepared by Ferdi. We will replace better one in future. See II-F in Ref.I.  
  : 1 logical  
  : 1 real (Ry). Used by hsfp0 (and hsfp0.sc for QSGW).  
  Poles of the Green function $G^{\rm LDA}$ are treated as if they have width esmr in hsfp0. If GaussSmear is on, each pole of $G^{\rm LDA}$ is smeared by a Gaussian function with esmr in the calculation of hsfp0. If GaussSmear is off, we assume rectangular smearing for the poles. Usually it is necessary to take rather smaller value than band gap for insulators. Try to use 0.003 or so in the case of Si and GaussSmear=on.  
  For metal, this esmr is somehow related to how we capture the Fermi surface; In principle, we have to take the limit n1n2n3 and esmr ). However, we may inevitably use some finite esmr to make calculations converged.)
* : 1 real (a.u.) only for one-shot case.
* deltaw is the interval for the numerical derivative in EQ.8. We calculate $\langle \psi^{{\bf k}n} |\Sigma(\epsilon^{{\bf k}n}+{\tt deltaw}) |\psi^{{\bf k}n} \rangle$ and $\langle \psi^{{\bf k}n} |\Sigma(\psi^{{\bf k}n}\!-\!{\tt deltaw})|\psi^{{\bf k}n} \rangle$ in addition to $\langle \psi^{{\bf k}n} |\Sigma(\epsilon^{{\bf k}n})|\psi^{{\bf k}n} \rangle$. From these values, we can calculate two (or second-derivative of ), as shown in SECU. It will help to see whether the used is O.K. or not.

1. Offset-gamma point.  
   :1 integer  
   Q0P gives how to determine the offset gamma points. Initially we take them as  
   1: |q| is ten times smaller than regular mesh.(default)  
   2: |q| is average in the Gamma cell (cell of BZ including Gamma point).  
   Then we choose only inequivalent ${\bf q}$ points based on the point group symmetry. Obtained offset gamma points is given in Q0P file.

* : 1 real (a.u.)  
  alpha corresponds to in EQ.48. alpha=1d0 is usually good in the sense that it seems to be almost a limit at . So you can usually fix it as alpha=1d0, and check the convergence as for n1n2n3.

1. core orthogonalization (default=off)  
   1 logical — recently, this option is not maintained — Better to use local orbital instead, so that core charge do not spill out.  
   If this is on, we enforce cores orthogonalized to valence and (these appear in II-C in Ref.I). This procedure enforce the correct orthogonal condition, thus we have correct behavior for the dielectric function at ${\bf q} \to 0$. However, it may deform core functions too much, especially in the case of shallow 3d (or maybe 4d) cores. So we don’t recommend use this option, even though then the orthogonality condition is somehow broken. Anyway you can check weather it affects to results or not by this switch.
2. QP self-consistent GW.  
   1 integer (no default).  
   This is required for QSGW calculation by the script gwsc. We have some possible ways to make GW self-consistent (how we determine $VUUUUUU{\rm xc}$ from calculated ). We now mostly use iSigMode=3.

* 3: Use $\displaystyle{\rm Re}\frac{\SigmaUUUUUU{nn'}(\epsilonUUUUUUn)+\SigmaUUUUUU{nn'}(\epsilonUUUUUU{n'})}{2}$ (mode-A in ).
* 1: Use $\displaystyle\SigmaUUUUUU{nn'}(EUUUUUUF) + \deltaUUUUUU{nn'}(\SigmaUUUUUU{nn'}(\epsilonUUUUUUn)-\SigmaUUUUUU{nn'}(EUUUUUUF))$ (mode-B in ).
* 5: Use $\displaystyle\deltaUUUUUU{nn'} \SigmaUUUUUU{nn}(\epsilonUUUUUUn)$ (Eigenvalue-only self-consistency, keeping the eigenfuncions as given)).
* See /gwsrc/sxcfUUUUUUfal2.sc.F@, which is called from the main routine hsfp0 (this is the routine to calculate self-energy)).

1. Others  
   1 logical (default=on)  
   These are for memory usage. When KeepEigen is on, eigenfunctions (Eigen) are kept in memory during calculation. If you have not enough memory in your machine, use them off. Then you can save memory usage. However, then we may have too frequent access to files. So %CPU might get lower. Be careful to use these options.

* 1 integer (default=0) If 0, it gives minimum standard output. If 40 or higher, it shows too much output. (these verbosity control is not well-organized yet).

1. ,, are for test purpose.xxx

## QPNT section

(only for one-shot GW. Not suitable to make band plot in BZ.) This section is to specify the q points and bands index for which you calculate the QP energies (QPE). An example is

=2.6mm

<QPNT>  
 --- Specify the q and band indeces, for which we evaluate the self-energy ---  
  
\*\*\* all q -->1, otherwise 0; up only -->1, otherwise 0  
 0 0  
\*\*\* no. states and band index for calculation.  
 3  
 15 16 17   
\*\*\* q-points, which shoud be in qbz. See KPNTin1BZ.  
 2  
 1 0.0000000000000000 0.0000000000000000 0.0000000000000000  
 2 0.2886751358562925 -0.5000000000000000 0.0000000000000000  
 3 0.0000000000000000 0.0000000000000000 0.3073140749846343  
</QPNT>

Numbers are read by free format read(5,\*), thus the numbers should be separated by space. At the next line to the first \*\*, you have to give two numbers used as flags. Both of them takes 0 or 1. 1st one is whether you calculate QPE for all q points (in IBZ) or not. If it is 1, you calculate QPE for all q. If it is 0, you calculate them only for q points specified within this file. In the case of metal where you want to calculate the Fermi energy for QPE, you need to calculate all the eigenvalues somehow above the Fermi energy (If you put 1, it is safer but too time-consuming). The second number is whether you calculate QPE for both spins or not. It is usually 0. In the case of antiferro material, it should be 1.

From the next line to the second \*\*, you have to specify the states for which you calculate the QPE. In this example, you calculate the 3 bands of QPE for 15th, 16th, and 17th eigenfunctions (they are ordered from the bottom).

From the next line to the third \*\*, you have to specify the q points. The first numbers of each line are dummy. In this case, you calculate QPE for two q points. The third q point is neglected because 2 is given at first.

When you generate GWinput.tmp, you see all the possible points are listed (these q points should be a part of the regular mesh points).

In the QSmode (gwsc), this section is neglected (then we calculate all QPE on regular mesh points); so its hsfp0part is quite expensive (usually it takes time more than hx0fp0).

Additional Note ————–  
num1 num2 (two integers).  
This override setting in <QPNT>@. (I think this switch may still work, but not checked recently).  
on (default is off)  
If this is on, you can specify any Q point which is not on the mesh point. For the purpose, we need to prepare eigenfunctions at extra $\bf k$ points. But it is automatic. In order to make the computation efficient. Even in this case, from the computational view, it is better to choose ${\bf q}$ on the two times finer divided mesh (or three times finer divided ${\bf k}$ mesh). This is used for Fig.6 in Phys. Rev. B 74, 245125 (2006).

## set QPNT for eps mode (QforEPS section)

For eps modes (scripts @, which are for linear responses. See Sec.[16](#linearr)), you have to specify q point in the following ways.  
1. on  
Then all Q point in IBZ are used.  
2. Use section as

<QforEPS>  
0d0 0d0 0.01d0  
0d0 0d0 0.02d0  
0d0 0d0 0.04d0  
0d0 0d0 0.08d0  
</QforEPS>

In addition, you can specify Q points as

<QforEPSL>  
0d0 0d0 0d0 1d0 0d0 0d0 8  
0d0 0d0 0d0 .5d0 .5d0 0d0 8  
</QforEPSL>

This is along the line— 8 point along the line (not left-end q; so omitting 0 0 0). The first line means line (0d0 0d0 0d0)—(1d0 0d0 0d0) is divided to 8. So we have 7 points, (0.125 0 0), (0.25 0 0),... (1 0 0).

## PRODUCT section

This section is to define product basis to expand and so. Numbers are read by free format read(5,\*),thus the numbers should be separated by space. The line number in this section is meaningful (you can not add comment lines).

width 15cm

=2.6mm

<PRODUCTUUUUUUBASIS>  
 tolerance to remove products due to poor linear-independency  
 0.100000D-04 ! =tolopt; larger gives smaller num. of product basis. See lbas and lbasC, which are output of hbasfp0.  
 lcutmx(atom) = maximum l-cutoff for the product basis. =4 is required for atoms with valence d, like Ni Ga  
 4 3  
 atom l nnvv nnc ! nnvv: num. of radial functions (valence) on the augmentation-waves, nnc: num. for core.  
 1 0 2 3  
 1 1 2 2  
 1 2 3 0  
 1 3 2 0  
 1 4 2 0  
 2 0 2 1  
 2 1 2 0  
 2 2 2 0  
 2 3 2 0  
 2 4 2 0  
 atom l n occ unocc ! Valence(1=yes,0=no)   
 1 0 1 1 1 ! 4SUUUUUUp -----   
 1 0 2 1 0 ! 4SUUUUUUd   
 1 1 1 1 1 ! 4PUUUUUUp   
 1 1 2 0 0 ! 4PUUUUUUd   
 1 2 1 1 1 ! 4DUUUUUUp   
 1 2 2 0 0 ! 4DUUUUUUd   
 1 2 3 1 1 ! 3DUUUUUUl   
 1 3 1 0 1 ! 4fUUUUUUp   
 1 3 2 0 0 ! 4fUUUUUUd   
 1 4 1 0 0 ! 5gUUUUUUp   
 1 4 2 0 0 ! 5gUUUUUUd   
 2 0 1 1 1 ! 2SUUUUUUp -----   
 2 0 2 0 0 ! 2SUUUUUUd   
 2 1 1 1 1 ! 2PUUUUUUp   
 2 1 2 0 0 ! 2PUUUUUUd   
 2 2 1 1 1 ! 3dUUUUUUp   
 2 2 2 0 0 ! 3dUUUUUUd   
 2 3 1 0 1 ! 4fUUUUUUp   
 2 3 2 0 0 ! 4fUUUUUUd   
 2 4 1 0 0 ! 5gUUUUUUp   
 2 4 2 0 0 ! 5gUUUUUUd   
 atom l n occ unocc ForX0 ForSxc ! Core (1=yes, 0=no)  
 1 0 1 0 0 0 0 ! 1S -----  
 1 0 2 0 0 0 0 ! 2S   
 1 0 3 0 0 0 0 ! 3S   
 1 1 1 0 0 0 0 ! 2P   
 1 1 2 0 0 0 0 ! 3P   
 2 0 1 0 0 0 0 ! 1S -----  
</PRODUCTUUUUUUBASIS>

width 15cm This section is read in the free format in fortran. So, e.g., 0.01@ works as same as 0.10000D-01@. The line order is important (you have to keep the order given by GWinput.tmp). Be careful atom atom id—lmf may re-order it and pass it to gw code. Look into LMTO file (generated by **mkGWIN2**); which contains crystal structure information after such re-ordering by lmf. I used !@ to make clear that things after !@ are comments. But !@ is not meaningful – just the expected numbers of data separated by blank(s) are read for each line from the beginning of lines.

* 0.100000D-02 ! =tolopt controls a number of Product basis to expand the Coulomb interaction within MTs. tolopt is a criterion to remove the poorly linear-independent product basis. Note that the product basis, which is to expand the Coulomb interaction, is different from the basis to expand eigenfunctions. In our experience, 0.100000D-02 (=0.001) is not so bad. If you like to reduce computational time use 0.01 or so, but a little dangerous in cases. With 0.0001, we can check stability on it.  
  (note: By supplying multiple numbers, we can specify tolopt atom by atom. Remember lmchk gives atom ID.)
* lcutmx(atom) is the l cutoff of product basis for atoms in the primitive cell (do lmchk for atom id). In the case of Oxygen, we can usually use lcutmx=2 (need check by the difference when you use lcutmx=2 or lcutmx=4). Then the computational time is reduced well.
* (dec2014:<PBASMAX>@ is not checked recently; see /main/hbasfp0.m.F@ and /gwsrc/basnfp.F@).) You can use <PBASMAX>@ section to override this setting. It is given as
* <PBASMAX>  
  1 5 5 5 3 3  
  2 5 5 3 2 3  
  3 3 3 2 2 2  
  </PBASMAX>
* The first number is for atom index (fixed), and other are product basis for each channel.
* The integer numbers in 4th line lcutmx gives the maximum angular momentum for the product basis for each atomic site. In our experience, lcutmx=4 is required when the semi-core (or valence ) electrons exist and we want to calculate the QP energies of them.
* Keep a block starting from " atom l nnvv nnc ..." as it originally generated in GWinput.tmp. It just shows that how many kinds of radial functions for cores and valence electrons for each atom and l. nnvv=2 in the case of and ; nnvv=3 in the case to add the local orbital in addition.
* There are two blocks after the line "atom l n occ unocc :Valence(1=yes, 0=no)’ and after "atom l n occ unocc ForX0 ForSxc ! Core (1=yes, 0=no)’. These are used to choose atomic basis to construct the product basis. The product basis are generated from the products of two atomic basis.
* GWinput.tmp generated by **mkGWIN2** contains labels on each orbitals as 4SUUUUUUp@, 4SUUUUUUd@, 4PUUUUUUp@... Here 4SUUUUUUp@ is for $\phiUUUUUU{4s}$; 4SUUUUUUd@ for ; 3DUUUUUUl@ for $\phi^{\rm local}UUUUUU{3d}$. Capital letter just after the principle-quantum number means the orbital is used as ‘Head of MTO’; lowercase means just used only as the ‘tail of MTO’.
* The switches for columns labeled as and . take 0 (not included) or 1 (included). With the switch, we can construct two groups of orbitals, and . In this sample GWIN2 as for atom 1, $\{ \phiUUUUUU{4s},\dot{\phi}UUUUUU{4s},\phiUUUUUU{4p},\phiUUUUUU{4d},\phi^{\rm local}UUUUUU{3d},
  \phi^{\rm core}UUUUUU{3s},\phi^{\rm core}UUUUUU{3p} \}$ consist the group , and $\{ \phiUUUUUU{4s},\phiUUUUUU{4p},\phiUUUUUU{4d},\phi^{\rm local}UUUUUU{3d},\phiUUUUUU{4f} \}$ consists the group . So the any product of combinations $\{ \phiUUUUUU{4s},\dot{\phi}UUUUUU{4s},\phiUUUUUU{4p},\phiUUUUUU{4d},\phi^{\rm local}UUUUUU{3d},
  \phi^{\rm core}UUUUUU{3s},\phi^{\rm core}UUUUUU{3p} \}
  \times \{ \phiUUUUUU{4s},\phiUUUUUU{4p},\phiUUUUUU{4d},\phi^{\rm local}UUUUUU{3d},\phiUUUUUU{4f} \}$ are included as for the basis of the product basis. As for atom 2, $\{ \phiUUUUUU{2s},\phiUUUUUU{2p},\phiUUUUUU{3d} \}
  \times \{ \phiUUUUUU{2s},\phiUUUUUU{2p},\phiUUUUUU{3d},\phiUUUUUU{4f} \}$ are included.
* Core section: (not worth to read, since we currently use no CORE2, =B=C=0@.)  
  Each line of the last section of Product BASIS forms
* =2.6mm
* atom l n occ unocc ForX0 ForSxc :CoreState(1=yes, 0=no)  
   1 2 1 A x B C
* At first you have to understand the concept of CORE1 and CORE2 in EQ.35 Ref.I. However, in our recent calculations, we do not use “CORE2” generally. So, in such a case, set =B=C=0@. And treat shallow cores (above Efermi2Ry or so ) as valence electron by “local orbital method” in lmf.
* Be careful. Current version is inconvenient... Need to repeat mkGWIN2 to generate GWinput template when you add PZ (local orbital).

[( Note: you can skip here if you don’t use CORE2.)

Each of A,x,B,C takes 0 or 1. There are some possible combination of these switches;

1. If you take ( A x B C )= (1 0 1 1), then the core is included in core2. In other words, this core is treated in the same manner of the valence electron.
2. If you take ( A x B C )= (0 0 0 0), then the core is included in core1. The (exchange only) self-energy related to this core is included in SEXcore.

* C is the key switch which determine whether it is included in core1 or core2. There could be another option.

1. If you take ( A x B C )= (1 0 0 1). This core is in core2. But it is not included in the calculation of and . This core is only included for SEX and SEC calculations.

These three kinds of choices are reasonable ones but we can consider some another choice. In the following, we show how these switches (,B,C@) affect executions called from gw (essentially as same as gw).

* hbasfp0(mode 3) :Product basis for exchange due to core.  
  We include the C=0 cores as a part of the product basis as if A=1 x=0.
* hsfp0(mode 3): exchange mode for core.  
  $\SigmaUUUUUU{\rm x}$ only due to the C=0 cores are calculated.
* hbasfp0 (mode1): Product basis.  
  Only see the switch A and x. The product basis is generated from (occupied unoccupied), where A=1 core is included as one of the occupied basis.
* hsfp0 (mode 1): exchange mode.  
  Only see the switch C. $\SigmaUUUUUU{\rm x}$ due to valence and due to C=1 cores are calculated.
* hx0fp (mode 1): calculation.  
  Only see the switch B. is calculates using all the valence and B=1 cores.
* hsfp0 (mode 2): correlation mode.  
  Only see the switch C. $\SigmaUUUUUU{\rm c}$ due to valence and due to C=1 are calculated.

After you perform or anything, you find output files lbas by hbasfp0 (mode1), and/or lbasc by hbasfp0 (mode3) for core. These contains important information about how many and how product basis are chosen. E.g. ’nbloch lbas@’ shows how many product basis are used in the calculations.

## ANFcond (we can skip here since we do not check this option now. Need fix this if necessary.

This file is used in **hx0fp0** in the calculation of (or rather in the program) to specify the antiferro condition.  
**Note** : Now only for the case that (a translation vector + spin flip) is a symmetry operation.  
This should be given by hand. For the cases of not antiferro, this file should not exist. Even if ANFcond does not exists for antiferro case, **hx0fp0** works but it requires about two time computational efforts.

=2.6mm

The existence of this file means the Antiferro condition is used for x0k  
Product basis B({\bf r}-{\bf a}) is translated to B({\bf r}-{\bf a}-Af})= B({\bf r}-{\bf a}'-TUUUUUU0})  
 1d0 1d0 1d0 ! Af=Antiferro translation vector in Cartesian.  
 1 2  
 2 1  
 3 4  
 4 3

The first line specifies the Antiferro translation vector. From the second line, we specify that atom in the primitive cell is mapped to what atom in the cell with the opposite spin by the translation. In this case, . You have to be careful as for the true atomic position used in the GW calculations can be different from the given atomic positions in ctrl.MnO. The true atomic positions is written in LMTO.

In the case of one-shot GW (gwand gw), it may be better to set "up only" QPE, so that you only calculate QPE of up spins at the same time.

In the case of gwsc, we just calculate QPE for up spins automatically (QPNT section is neglected).

# Main Output Files of GW part

## QPU

This is the main output[[13]](#footnote-97) in human readable format.

An example of one-shot GW by gw) is (In the case of QSGW, () is not shown):

=2.6mm

===============================================================  
 quasiparticle energies MAJORITY  
 ===============================================================  
EUUUUUUshift= 0.4263273221017709D+00 0.6075150850568627D+00 0.7046628446164018D+00 eV  
  
 q state SEx SExcore SEc vxc dSE dSEnoZ eLDA eQP eQPnoZ eHF Z 2Z\*Simg ReS(elda)  
0.0 0.0 0.0 1 -29.56 -1.97 10.40 -20.22 -0.52 -0.90 -19.08 -19.42 -19.71 -30.81 0.58 0.95 -21.12  
0.0 0.0 0.0 2 -30.52 -2.24 10.09 -21.53 -0.70 -1.14 -18.06 -18.58 -18.93 -29.72 0.61 0.96 -22.66  
0.0 0.0 0.0 3 -20.67 -1.87 5.97 -16.85 0.19 0.28 -7.20 -6.83 -6.65 -13.32 0.67 0.66 -16.57  
 ...

From the 6h line, we have the eigenvalue data. All of the unit of energy is in eV. We should note that the zero-level of these values eLDA eQP eQPnoZ can be changed by hqpe. This eLDA - E are the eigenvalues relative to a Fermi energy determined by the smearing method. Detailed value of eLDA is in TOTE2.UP. Detailed value of eLDA- E is in TOTE.UP.

q : ${\bf k}$ vector

state: Band index , which is from the lowest eigenvalue (not include cores).

SEx: = $= \langle{\bf k}n{{\bf k}n}|\SigmaUUUUUU{\rm x}^{\rm core2+valence}({\bf r},{\bf r}^{\prime})|{\bf k}n{{\bf k}n}\rangle$

SExcore: $= \langle{\bf k}n{{\bf k}n}|\SigmaUUUUUU{\rm x}^{\rm core1}({\bf r},{\bf r}^{\prime})|{\bf k}n{{\bf k}n}\rangle$

SEc: $= \langle{\bf k}n{{\bf k}n}|\SigmaUUUUUU{\rm c}^{\rm core2+valence}({\bf r},{\bf r}^{\prime},\epsilonUUUUUUn({\bf k}))|{\bf k}n{{\bf k}n}\rangle$

vxc: LDA exchange correlation energy. $\langle{\bf k}n{{\bf k}n}|VUUUUUU{\rm xc}^{\rm LDA}([nUUUUUU{\rm total}],{\bf r})|{\bf k}n{{\bf k}n}\rangle$

dSE: $ZUUUUUU{n{\bf k}}\times$ dSEnoZ

dSEnoZ: $\langle{\bf k}n{{\bf k}n}|
\SigmaUUUUUU{\rm x}^{\rm core1}({\bf r},{\bf r}^{\prime})+
\SigmaUUUUUU{\rm xc}^{\rm core2+valence}({\bf r},{\bf r}^{\prime},\epsilonUUUUUUn({\bf k}))
|{\bf k}n{{\bf k}n}\rangle
- \langle{\bf k}n{{\bf k}n}|VUUUUUU{\rm xc}^{\rm LDA}([nUUUUUU{\rm total}],{\bf r})|{\bf k}n{{\bf k}n}\rangle$

= SEx + SExcore + SEc - vxc

eLDA: LDA eigenvalues. $\epsilonUUUUUUn({\bf k})$

eQP: QP energy. $\epsilonUUUUUUn({\bf k})$+dSE

eQPnoZ: QP energy without . $\epsilonUUUUUUn({\bf k})$+dSEnoZ

eHF: HF energy of 1st iteration. $\epsilonUUUUUUn({\bf k})$+SEx + SExcore -vxc

Z: Z factor. $ZUUUUUU{n{\bf k}}$

2Z\*Simg: Quasi-particle life time. $2 ZUUUUUU{n{\bf k}} \times {\rm Im}
\langle{\bf k}n{{\bf k}n}|
\SigmaUUUUUU{\rm c}^{\rm core2+valence}({\bf r},{\bf r}^{\prime},\epsilonUUUUUUn({\bf k}))
|{\bf k}n{{\bf k}n}\rangle$

(Is this really the usual definition of the life time?—don’t believe me)

ReS(elda): ${\rm Re}
\langle{\bf k}n{{\bf k}n}|
\SigmaUUUUUU{\rm x}^{\rm core1}({\bf r},{\bf r}^{\prime})+
\SigmaUUUUUU{\rm xc}^{\rm core2+valence}({\bf r},{\bf r}^{\prime},\epsilonUUUUUUn({\bf k}))
|{\bf k}n{{\bf k}n}\rangle$

## XCU

LDA exchange-correlation. Detailed data of above vxc.

## SEXU

Exchange part of the self-energy due to valence electrons. Detailed data of above SEx.

## SEXcoreU

Exchange part of the self-energy due to core. Detailed data of above SExcore.

## SECU

Correlation part of the self-energy. Detailed data of above SEc.

## TOTE.UP (TOTE.DN)

This is a central output. It contains LDA and QP energies. These values are relative to a Fermi energy determined by the smearing method. It contains two kind of QP energies QP QPnoZ. The first line contains the Fermi energy in Ry determined by the smearing method. It is also shown in the end of DOSACC.lda.

## TOTE2.UP (TOTE2.DN)

This is a central output. It contains zero-level shifts from TOTE.UP. The first line contains the Fermi energy in eV (= the Fermi energy in TOTE.UP but it is in Ry) and three energy shifts E, which are the same values in the 4th line of QPU.

Note that all .chk files are just to check calculations (not read in by successive executions).

## DOSACC.lda

This lists all the eigenvalues in ascendant order. States with almost the same eigenvalues are degenerated states. The 4th column contains number of electrons up to the eigenvalue.

## DOSACC2.lda

This is similar with DOSACC.lda. But we remove the degeneracy.

## Core.chk

Used core eigenfunctions.

## VXCFP.chk

This contains eigenvalues and $\langle \psiUUUUUU{{\bf k}n} |VUUUUUU{\rm xc} |\psiUUUUUU{{\bf k}n}\rangle$ in both units, Ry and eV. See below.

## The Fermi energies in this code.

We mainly have two kinds of Fermi energy $EUUUUUU{\rm FEERMI}^{\rm smear}$ $EUUUUUU{\rm FEERMI}^{\rm tetra}$.

1. At first eigenvalues given by lmfgw is in VXCFP.chk. You can see

* %head VXCFP.chk  
  ### LDA exchange correlation ###  
  # qvec ikp iband eigen VXC(ntotal) VXC(nvalence) eigen(eV) VXC(ntotal)(eV) VXC(nvalence)  
   0.0000 0.0000 0.0000 1 1 -0.96932423 -1.00727912 0.00000000 -13.18843159 -13.70483831 0.00000000  
  ...
* These are raw values. TOTE contains the eigenvalues but relative to a Fermi energy $EUUUUUU{\rm FEERMI}^{\rm smear}$ which is determined by the smearing method. It is also shown at the top part of output files lsx and lsc. And you also see the value at the end of DOSACC.lda.  
  This is the head of TOTE.UP;
* %head TOTE.UP  
   43 8 8.520283353474250E-003  
   0.0000000 0.0000000 0.0000000 1 1 -0.1330435686590073D+02 -0.1322984339282777D+02 -0.1319228487875673D+02 0.6648715256312900D+00  
   0.0000000 0.0000000 0.0000000 2 1 -0.7555264915356062D+00 -0.6267595395613325D+00 -0.6009483309649021D+00 0.8330216344848770D+00  
  ...
* Here $EUUUUUU{\rm FEERMI}^{\rm smear}$=8.520283353474250E-003. From the second lines, they are LDA eigenvalues and QP energies (Z included and Z=1); they are relative to the $EUUUUUU{\rm FEERMI}^{\rm smear}$.  
  -13.18843159 eV - $EUUUUUU{\rm FEERMI}^{\rm smear}$(which should be translated into in eV) = -0.1330435686590073D+02 eV. Here -13.18843159 is the value in VXCFP.chk shown above.

1. There is the another Fermi energy $EUUUUUU{\rm FEERMI}^{\rm tetra}$, which is used by mode 11 (or mode 1) of hx0fp0 in **gw**. It is determined by heftet and stored in EFERMI.
2. hqpe gives TOTE2.UP and QPU. They contains the same values. You can see eQP eQPnoZ Z@ not only in QPU but also in TOTE2.UP. At top lines of TOTE2.UP, you see

* %head TOTE2.UP  
   43 8 0.1159252712507000D+00 0.7555207081466229D+00 0.6267572296579150D+00 0.6009467172357787D+00  
   0.0000000 0.0000000 0.0000000 1 1 -0.1254883615775411D+02 -0.1260308616316985D+02 -0.1259133816152095D+02 0.6648715256312900D+00  
   0.0000000 0.0000000 0.0000000 2 1 -0.5783388983382487D-05 -0.2309903417430093D-05 -0.1613729123328689D-05 0.8330216344848770D+00  
   0.0000000 0.0000000 0.0000000 3 1 -0.1369098933889923D-05 -0.6195200397129952D-07 0.2047424042528334D-06 0.8330216088297998D+00  
   0.0000000 0.0000000 0.0000000 4 1 0.0000000000000000D+00 0.0000000000000000D+00 0.0000000000000000D+00 0.8330216340404680D+00  
  ..
* ,where a number in first line $EUUUUUU{\rm FEERMI}^{\rm smear}$= 0.1159252712507000D+00 eV =8.520283353474250E-003 Ry, the same as the previous one. This is a case when you did hqpe with augment 4 (it means we set the 4th-band eigenvalue zero). Another 3 values in the first line are shifts from TOTE. Shown eshift(eLDA) = 0.7555207081466229D+00 eV. E.g., the second line shows  
  -0.1254883615775411D+02 eV= -0.1330435686590073D+02(in TOTE) + eshift(eLDA) eV.  
  When you do hqpemetal, three shifts at the first line in TOTE2.UP is determined so as to give the eigenvalues relative to the Fermi energies shown in EFERMI, EFERMI.QP1, and EFERMI.QPz=1. These are Fermi energies by tetrahedron method.

As for **gwband**, it recalculates eigenvalues for all **q** along SYML. Then  
the default "zerolevel" = $EUUUUUU{\rm FEERMI}^{\rm smear}$ - eshift(lda). Because the eigenvalues given by this bandmode are presumably the same, we have  
Shown LDA eigenvalue  
= -13.18843159(raw data by band mode—same as that in VXCFP.chk) - zerolevel  
= (-13.18843159 - EFERMIsmear) + eshift(lda).  
= -0.1330435686590073D+02( this is in TOTE.UP) + eshift(lda)  
= -0.1254883615775411D+02( this is in TOTE2.UP).  
It means that values in TOTE2.UP recovers. But if raw data by band mode is different from it, these is a trouble. It does not recover the values in TOTE2.UP(=QPU).

As for the QPE, we calculate the difference from LDA values in TOTE2.UP at first, and add the difference to the Shown LDA eigenvalue.

# mkGWIN2 and its I/O Files

(QPNT.chk contains irreducible k point for given n1 n2 n3; KPTin1BZ.gwinit.chk contain all k points in Brillouin Zone).

The purpose of the script **mkGWIN2** is to give a template GWinput.tmp. The script is complicated because of historical reasons. However, its essential is simple; we calls three executions in this script as  
echo 0 lmfgw si  
echo 1 gwinit  
echo qg4gw  
. We explain each by each.

## **echo 0lmfgw**

— makes SYMOPS LATTC CLASS NLAindx.

[lmfgw]

: This is a file, which contains your supplied n1 n2 n3 when you invoke the script. This file is given within the script of mkGWIN2 (as "here document").

=2.8mm

cat <<EOF >GWIN0  
n1 n2 n3  
 $n1 $n2 $n3  
cut  
 4.0 3.0  
alpha  
 1  
Number of bands  
 999 99999.0   
 999 3.0   
EOF

: Master input file of lmf calculation.

: contains the information of primitive translation vectors, lmxa and konf. See [[lattc]](#lattc)

: The point group operations. See [[symops]](#symops)

: Equivalent atomic positions are called as ’class’. This small file contains a map between atomic site and ’class’.

: This file contains indexes ($pUUUUUU{\rm valence}, l, a$ ) for orbitals in the MT. ($pUUUUUU{\rm valence}$ is radial function index, is atomic site index). Eigenfunctions are expanded in this order.

: Number of MTOs for each atomic site. (this is used only from hqpe—QSGW mode).

: not meaningful at this stage

: not meaningful at this stage

## **gwinit**

— Get GWIN2.tmp and QPNT.tmp

:

:

:

:

: A part of GWinput.tmp

: A part of GWinput.tmp

: check KPNT in the 1BZ.

If SYML exist, **gwinit** gives also a template QPNTforSYML.tmp suitable for such SYML. Here SYML specify how to plot the energy band. See explanation for **bandplot** script.

Note that LATTC SYMOPS CLASS NLAindx are overwritten when you execute **gw** because we repeat **echo 0lmf** at the head of **gw**.

## **echo -100qg4gw**

— Generate GWinput.tmp

: (copy of GWIN0.tmp by **gwinit**)

: (copy of GWIN2.tmp by **gwinit**)

: (copy of QPNT.tmp by **gwinit**)

: (this is copied to GWinput.tmp)  
This command "echo -100qg4gw" is a file converter from these two files into GWinput. And it is copied to GWinput.tmp. (mkGIN keeps GWinput if it exist before you invoke it.).

# gwsc script and its I/O Files

In gwsc, we have a loop of QSGW self-consistency. Look into the gwsc script. In each iteration, we perform these fortran programs;

=2.8mm

NOUUUUUUMPI=0 #this is used for non-mpi versions of fortran program.  
  
 ### self-consistent calculation for given Sigma(self-energy) ###  
  
 runUUUUUUarg '---' $MPIUUUUUUSIZE $nfpgw /lmf-MPIK llmf $TARGET  
  
  
 ### Preparation stage ##############################################  
  
 argin=0; runUUUUUUarg $argin $NOUUUUUUMPI $nfpgw /lmfgw llmfgw00 $TARGET   
 argin=1; runUUUUUUarg $argin $NOUUUUUUMPI $nfpgw /qg4gw lqg4gw #Generate requied q+G v  
 argin=1; runUUUUUUarg $argin $MPIUUUUUUSIZE $nfpgw /lmfgw-MPIK llmfgw01 $TARGET  
 runUUUUUUarg '---' $NOUUUUUUMPI $nfpgw /lmf2gw llmf2gw #reform data for gw  
  
 ### Main stage of gw ################################################  
  
 argin=0; runUUUUUUarg $argin $NOUUUUUUMPI $nfpgw /rdata4gwUUUUUUv2 lrdata4gwUUUUUUv2 #prepare files  
 argin=1; runUUUUUUarg $argin $NOUUUUUUMPI $nfpgw /heftet leftet # A file EFERMI for hx0fp0  
 argin=1; runUUUUUUarg $argin $NOUUUUUUMPI $nfpgw /hchknw lchknw # A file NW, containing nw   
  
 ## Core part of the self-energy (exchange only) ##  
  
 argin=3; runUUUUUUarg $argin $NOUUUUUUMPI $nfpgw /hbasfp0 lbasC # Product basis generation   
 argin=3; runUUUUUUarg $argin $MPIUUUUUUSIZE $nfpgw /hvccfp0 lvccC # Coulomb matrix for lbasC   
 argin=3; runUUUUUUarg $argin $MPIUUUUUUSIZE $nfpgw /hsfp0UUUUUUsc lsxC # Sigma from core1  
  
  
 ## Valence part of the self-energy Sigma ##  
  
 argin=0; runUUUUUUarg $argin $NOUUUUUUMPI $nfpgw /hbasfp0 lbas # Product basis generation   
 argin=0; runUUUUUUarg $argin $MPIUUUUUUSIZE $nfpgw /hvccfp0 lvcc # Coulomb matrix for lbas   
 argin=1; runUUUUUUarg $argin $MPIUUUUUUSIZE $nfpgw /hsfp0UUUUUUsc lsx # Exchange Sigma  
 argin=11; runUUUUUUarg $argin $MPIUUUUUUSIZE $nfpgw /hx0fp0UUUUUUsc lx0 $lx0UUUUUUparaUUUUUUoption #x0 part  
 argin=2; runUUUUUUarg $argin $MPIUUUUUUSIZE $nfpgw /hsfp0UUUUUUsc lsc #correlation Sigma  
 argin=0; runUUUUUUarg $argin $NOUUUUUUMPI $nfpgw /hqpeUUUUUUsc lqpe #all Sigma are combined.

**run**:  
Here a subroutine of bash **run** was used, which is given in ecalj/lm7K; it just invoke a command with the argument $argin@ (this is read by
read(\*,\*) in fortran). In cases with \verb@MPIUUUUUUSIZE@$/=0$,
mpirun is invoked. Console out
put go to l\* files. For example,
%\begin{quote}
%\verb@argin=2; runUUUUUUarg $argin $MPIUUUUUUSIZE $nfpgw /hsfp0UUUUUUsc lsc #correlation Sigma@
%\end{quote}
invokes {\tt hsfp0\UUUUUUsc} with argument '2' by mpirun with the -np
\$MPI\UUUUUUSIZE. Console outputs are written into logfiles such as {\sf lqpe}.
\verb@$nfpgw@ contains path to the execution binaries.  
   
In the following, We explain input/output files for each fortran program. Note that “echo 0lmfgw” means invoking lmfgw with =0@.

## **echo 0 lmfgw si**

See Sec.[[lmfgw]](#lmfgw).

## **echo 1 qg4gw**

This makes **q** points, and **G** vectors for these **q**. (**q** was **k** in previous sections.) Main routine of qg4gw is fpgw/main/qg4gw.m.F and calls fpgw/gwsrc/mkqg.F

:

:

:

: (bin) q and G vector for the eigenfunctions.

: (bin) q and G vector for the Coulomb matrix

: offset- points which are the replacement of the q=0 points. See section[[xxx]](#xxx).

: q points in the Irreducible BZ.

: (bin) BZ data for integration (include tetrahedrons if necessary). See e.g. main/hx0fp0.sc.F and search "call read", which is a readin routine of this file defined in rwbzdata.F.

: list of q in the 1st BZ for check.

: q point in the 1st BZ.

: Required information for the BZ integration (mainly in order to evaluate the weight in the cell). See Eq.xxx in .

## **echo 1lmfgw si**

Calculate eigenfunctions, eigenvalues and $\langle \psi | HUUUUUU{\rm KS}|\psi\rangle$

:

: (bin) Restart file of the lmf calculation. It contains all information

: (bin) If this exist and

: :

: :

: (bin) atomic data

: (bin) band data

: (bin) $\langle \psi|HUUUUUU{\rm KS}|\psi \rangle$

: (bin) $\langle \psi|HUUUUUU{\rm KS}-VUUUUUU{\rm xc}(nUUUUUU{\rm total})|\psi \rangle$.

: (bin) used in hqpe.sc.m.f as "v" and "evec").  
vxc.si contains $\langle \psi|VUUUUUU{\rm xc}(nUUUUUU{\rm total})|\psi \rangle$ including off-diagonal part. evec.si contains eigenfunctions.

: norm check (only for check) This is like this

=2.8mm

> head -20 normchk.si  
# IPW IPW(diag) Onsite(tot) Onsite(phi) Total  
 0.436015 0.805123 0.563972 0.562573 0.999988  
 0.339134 0.620353 0.660515 0.656881 0.999649  
 0.339133 0.620353 0.660516 0.656882 0.999649  
 0.339133 0.620353 0.660516 0.656882 0.999649  
 0.507738 0.648515 0.492040 0.487673 0.999778  
...

This check is sometimes important for debugging and to determine the cutoff parameter . The first line (corresponding to 1st band of 1st q point) means that total normalization almost unity = 0.999988 = 0.436015 + 0.563972. Because we expand the MTO by IPW, the normalization is a bit different from unity, especially for higher bands. You can see that it get closer to unity for larger QGcut, though it does not reach to unity because of some contribution of the higher angular momentum contribution within MT. [Values of (phi)@ are not correctly shown in the case when you use local orbital.]

Due to historical reason, data in vxc.si and exec.si and others contains duplicated data.

## **lmf2gw**

All the required information are stored into DATA4GW2 and CphiGeig.

:

:

:

:

:

:

:

: (bin) Main data for GW calculations.

I/O of DATA4GW2 is controlled by gwinput.f, which contains detailed information.

: (bin) Eigenfunctions for GW calculations.

: Eigenvalue and Vxc check (only used for check)

It is like this;

=2.8mm

### LDA exchange correlation ###  
# qvec ikp iband eigen VXC(ntotal) VXC(nvalence) eigen(eV) VXC(ntotal)(eV) VXC(nvalence)  
0.0000 0.0000 0.0000 1 1 -0.68505346 -0.91850436 0.00000000 -9.32070032 -12.49698668 0.00000000  
0.0000 0.0000 0.0000 1 2 0.19292662 -0.99853478 0.00000000 2.62492096 -13.58586453 0.00000000  
0.0000 0.0000 0.0000 1 3 0.19292763 -0.99853469 0.00000000 2.62493477 -13.58586334 0.00000000  
0.0000 0.0000 0.0000 1 4 0.19292777 -0.99853461 0.00000000 2.62493664 -13.58586222 0.00000000  
 ...

Here VXC(nvalence) is not used now. The eigenvalue in eigen is in Ry.

— This is the end of the preparation stage. —

From here, the main stage.

## **rdata4gw2**

— Read DATA4GW2 and some files, and decompose it into files required in the following steps. (checked! dec2014)

:

:

:

:

:

:

:

: points group operations.

: data size

: core eigenfunctions just for check.

: basic date for the crystal.

: (bin) valence eigen value

: core data and core eigenvalues

: (bin) Coefficients of eigenfunctions as for the atomic-like argumentation waves in MTs’.

: (bin) Coefficients of eigenfunctions as for IPW.

: (bin) All the radial functions.

: index set for CPHI. This is not refereed just a check write.

: index set for core. This is not refereed just a check write.

: (bin) this is for diagonal elements of $V^{\rm LDA}UUUUUU{\rm xc}(nUUUUUU{\rm total})$.

: (bin) Overlap matrix of IPW. xxxxxxxxxx

: (bin) PPOVLG Overlap matrix xxxxxxxxxxoxf IPW. not exactly the the overlap matrix. see around line 500 in rdata4gw.m.f

: (bin) xxxxxxxxxxxxx

: (bin) Required inputs for hvccfp0. Information in this files.

: q point info. Only used for parallel test mode.

: Norm check. These numbers should be almost the same as those in normchk.si

These files are input for the following steps. The name of file U means that it relates to up-spin. We have D files in the case of spin-polarized calculation with nspin=2.

## **echo 1heftet**

— Get the Fermi energy EFERMI by tetrahedron method. It is used in **hx0fp0**.

:

:

:

: (dummy)

: (dummy)

:

:

: contains Fermi energy given by the tetrahedron method. It is used in **hx0fp0** but not in **hsfp0**.

: They are lists of the all the eigenvalues from the bottom. DOSACC2.lda is a list to show only the un-degenerated eigenvalues. They are just check write. But it is an indicator for you to determine in GWinput.

## **hchknw**

— Calculate the required number of points along real axis.  
This NW is not essentially used in gw (but required as a dummy file). Only used in gw.

:

:

: (dummy)

: (dummy)

: contains number of points.

## **echo 3hbasfp0**

— Make product basis.

Mode 3 is for the core states. It generate a product basis on each MT suitable to expand to calculate the exchange part due to core. See explanations for the input file of GWinput.

:

:

:

: (bin) Product basis functions

: (bin) Radial integrals on each MT, symbolically written as

: Valence radial functions (for check).

## **echo 0 hvccfp0**

— Calculate the Coulomb matrix in the Mixed basis

:

:

:

: The Coulomb matrix expanded in the mixed basis

: This is used only for dielectric-constant calculation (mode 2 or 3 of **hx0fp0**). This contains the expansion of the plane wave $\exp( i{\bf q r})$ in the mixed basis. See Usuda’s note.

## **echo 3hsfp0**

— Exchange part of the self-energy for the core

:

:

:

:

:

:

:

:

:

: The core part of the exchange self-energy for **q** and band index specified in QPNT. See [[mainoutput]](#mainoutput).

## **echo 0hsfp0**

— Make product basis for the valence part.

## **echo 1hsfp0**

— Exchange part of the self-energy for the valence part.

: The LDA exchange self-energy for **q** and band index specified in QPNT. See [[mainoutput]](#mainoutput).

: The valence part of the exchange self-energy for **q** and band index specified in QPNT. See [[mainoutput]](#mainoutput).

## **echo 11hx0fp0**

— Screened Coulomb interaction (Sergey mode)

:

: dummy

:

:

:

:

:

:

: (optional) This file is to specify antiferro condition. This should not exist for other cases. This file should be given by hand.

: size of the dielectric function

: (bin) in the expansion of mixed basis along the real axis

: (bin) in the expansion of mixed basis along the imaginary axis

## **echo 12hsfp0**

— Correlation part of the self-energy(Sergey mode)

: These are readin by genallcf3.

:

:

Radial integrals on each MT, symbolically written as . These are generated by hbasfp0.

:

:

:

: The correlation part of the self-energy for **q** and band index specified in QPNT. See [[mainoutput]](#mainoutput).

## **echo 0hqpe**

— Summarize the output

: See [[mainoutput]](#mainoutput).

: The QP energies and related value summary in human interface. See [[mainoutput]](#mainoutput).

: The detailed values of the QP energies. See [[mainoutput]](#mainoutput).

: The detailed values of the QP energy. See [[mainoutput]](#mainoutput). This is used for **bndplot**.

NOTE: For example, if you do @, it just shift zero level of QPE, so that 4th line (counted from top) eigenvalue (in QPU) is to be zero.

# Check list for convergence on GW calculations

Results could be dependent on cutoff parameters in GWinput. In my opinion, generally speaking, it is so easy to have convergence more than 0.1eV for band gap...

* Number of k points .  
  Probably (or ) are reasonable choice for insulator in the case of two atoms such as GaAs. In other words, ‘’’used periodic cell volume” = (# of atoms) (Volume per atom)”. For example, for atoms per cell, we can use instead of (this is the same as in the case of LDA).
* For metals, or more per atom may be required. But it depends on case by case.
* MTO’s and APWs. Basis for eigenfunctions.
* If we like to get “best converged results, we may need to use large enough MTO’s. (I think the default setting is reasonable; but there is a room to change MTO settings in ctrl.\*.). But, in cases, we can not use large APW cutoff () more than Ry because of poor linear dependency of basis set (calculation in LDA level fails). In such a case, we need to use “smaller MT radius =@”. Then we may need semi-core as local orbital when “its spillout in the outside of MTs is too large”.
* In GWinput, we can set the number of unoccupied states which you take into account by , ,, and . But we usually unset them except for gwsc.
* (We may need a kind of completeness of the basis set; the completeness could be important from the view of ‘Coulomb hole’ picture.)
* NOTE: Current PMT-QSGW method expand the static version of self-energy of QSGW just in the basis of MTO’s (no APWs). Thus the expansion can be unsatisfactory (it depends on case by case, and required convergence). In such a case, we inevitably have to use empty spheres (MTO’s) which is for empty region.
* Cores.  
  We usually use all cores as **core1** (exchange only core). If necessary, it is better to treat shallow cores by the local orbitals (such cores are treated as valence).
* —followings are obsolate —
* If we treat cores by **core2** (not only for exchange, but also in dielectric functions), we have to be careful about the core wave orthogonalization with respect to valence eigenfuncitons; this is a little complicated; probably it is better not to use **core2**).
* (This is related to (only for core2). If it affects so much, function might be too poor due to the poor orthogonality condition between core and valence.) (NOTE: is not maintained recently)
* ————-
* IPW cutoff to expand eigenfuncitons in the interstitial region. We usually use 4.0@. Usually not so bad. Larger is better but expensive. You may test calculations with 3.0@, and how much difference of results.
* IPW cutoff to expand the Coulomb interaction in the interstitial region (Mixed product basis(MPB) consist of this IPW and PB). interstitial region. We usually use 3.0@. Usually not so bad. Larger is better but expensive. You may test calculations with 2.5@, and how much difference of results.
* Product basis section.  
  At least, =4 will be necessary for atoms with d electrons. (but =2 look reasonable for oxygen, =1 for hydrogen. Need check).
* .  
  In our experience, =0.003000@ (default) is reasonable. But there is a room to check stability on it for metals. In principle, for larger n1n2n3, we can use smaller .
* It will be worth to try to check how much the results changed due to them. But usually =0.005, omgUUUUUUc=0.04@ is not so bad a choice. As for =10@ seems to be not so bad usually, but it is safer to check the convergence on it (test cases with =6,10,12,16@).
* . a.u. will be not so bad. See two Z values shown in SXCU. It is better to try to check how about the dependence on this.
* . (on is default). If off, we use off-Gamma mesh (Gamma point is between mesh points) for dielectric functions when we perform GW or QSGW. “chioff” may accelerate convergence on number of k points.
* in ctrl This used fixed number of G vector determined at q=0

# Linear response calculations

With these scripts for linear response calculations, @, we can calculate ${\bf q}$-dependent dielectric funciton $\epsilon(\omega ,{\bf q})$ (and , ) (and for spin fluctuation). But (because of numerical reason), we can not use ${\bf q}= 0$ limit. (if $|{\bf q}|$ is too small, we have numerical problem, zero divided by zero, because we have not implemented the version to use ${\bf q}=0$.)

* eps  
  Dielectric function epsilon with local field correction. Expensive calculation (we may need to reduce number of wing parts in future...).
* epsPP  
  epsilon without local field correction. $1- \langle e^{i {\bf q}{\bf r}}|v|e^{i {\bf q}{\bf r}}\rangle \langle e^{i {\bf q}{\bf r}}| \left( \chi^{0} \right) |e^{i {\bf q}{\bf r}}\rangle$
* epsPP
* For spin susceptibility. This essentially calculate non-interacting spin susceptibility. Then it is used for the calculation of full spin susceptibility with /caljUUUUUU\*.F@ programs (small quick programs). See spin wave paper. See spin susceptibility section Sec.[[xxx]](#xxx).
* (not maintained now; we will recover this)eps
* This gives full non-interacting spin susceptibility. Testing. We have to determine (Stoner ) for the determination of full spin susceptibility. TDLDA? or so?
* (This is old mode — not maintained) epsPP
* For spin susceptibility, spin susceptibility In this script, You have to assign that isp=1 is majority, isp=2 is minority. This is with long wave approximation.

———————

We use the histogram method (the Hilbert transformation method); we first calculate its imaginary parts with the tetrahedron technique for dielectric functions. Then we get its real part by the Hilbert transformation.  
You need to choose . The width of histogram bins are getting larger when omega gets larger. dw is the size of histogram-bin width at omega=0. At omega=omg, its width gets twiced.

To plot dielectric function with reasonable resolution, it might be better to set 0.001@ and 0.1@ for example. You may have to choose small enough omega for spin wave mode as 0.001 Ry (Or smaller). omgis given like 0.05 Ry or so. But sometimes it can be like 1Ry.

epsPP only calculation an matrix element of dielectric funciton for $exp(i {\bf q}{\bf r})$. Thus very faster than eps mode. It uses a a special product basis set for cases without inversion (problem is in how to expand $\exp(i {\bf q}{\bf r})$ in the MPB; the product basis is not from phi and phidot, but from spherical Bessel functions).

In \*UUUUUUlmfhUUUUUU\* modes( I now use little for \*UUUUUUlmfUUUUUU\* modes), you can use small enough delta.  
Use small enough delta (=-1e-8 a.u.) for spin wave modes (also you can use it for   
dielectric function and GW). This is necessary because pole is too smeared   
if you use larger delta.

## eps, epsPP: the dielectric functions

You can invoke the script, e.g. as "eps  si".

—————-

Specify ${\bf q}$ point in <QforEPS>@ or so. Mesh for is specified by .

The obtained data are in EPS\*.dat and EPS\*.nlfc.dat. EPS\*.nlfc.dat contains the result without local-field correction EPS\*.dat contains the result with local-field correction (this is generated only for . Both of them contains

**q**(1:3), , Re() Im(), Re(1/), In(1/)  
in each line.

* This code works OK only for ${\bf q}$ is near 0. Be careful for ${\bf q} \to 0$ limit. Too small ${\bf q}$ can give strange spectrum at high energy (real part is affected by it)  
  Because ${\bf q} \to 0$ gives too large cancellation effects (the denominator and numerator go to zero—it means we need very accurate orthogonalization between occupied and unoccupied states). This is a kind of disadvantage of our method (though there is an advantage— our code can calculate dielectric function even for metal as far as you use large enough number of ${\bf k}$ point.)
* The calculate of dielectric functions usually requires so many point. For example, for Si, 1 n2 n3 = 4 4 4@ is too small. It gives too large dielectric constants though the converged value should be . (we need 10x10x10 or more like 20x20x20 for some reasonable results). For GaAs, we observed that reasonable requires rather large number of ${\bf q}$ points like 15x15x15 or 20x20x20 for . This is too time-consuming to get result (but you can use “very small product basis”(just sp polarization for this purpose; it makes speed up so much). Or, you can calculate " without LFC". See section for eps.
* Core orthogonalization problem (only when core2 is used)  
  —- is not maintained recently — gives so serious effect for , if you include some cores as "**core2**" in the product basis setting. (This means that you includes transitions from "**core2** to valence" in the calculation of ).
* Then you have to use "CoreOrth on". Without it, you will have rather large imaginary part at rather high energy Such transitions from core to higher valence bands is artificial due to the incomplete orthogonality between core and the higher bands. However, shallower semi-core might be deformed too much by this option. Try to plot Core.chk files, which contains core radial functions. Anyway, it is better to treat shallow core as valence by “local orbital”.

## epsPP: the dielectric function(No LFC— faster)

You can calculate without LFC by **epsPP**. It is very faster than eps.

To calculate $\epsilon({\bf q},\omega)$ without LFC accurately, the best basis set for the expansion of the Coulomb matrix within MT is apparently not the product basis, but the Bessel functions corresponding to the plane waves $\exp(i{\bf q r})$. We use such a basis in this mode. However, our experience shows that the changes are little even with the usual product basis (we don’t describe this here).

## How to calculate correct dielectric funciton?

(this subsection is essentially OK... but need to clean it up. dec2014)

There are prolems to calculate correct epsilon.  
At first, we talk about epsPPUUUUUUlmfh, which is No LFC. Main problem are   
  
-----------------  
1.Convergence for number of k point(specified by n1n2n3).   
 Roughly speaking, 20x20x20 is required for not-so-bad results for Fe and Ni.  
 It is better to do 30x30x30 to see convergence check.  
 However, in the case of ZB-MnAs (maybe because of simple structure around Ef),  
 it requires less q points.  
  
 figs are for GaAs.  
 fig001: n1n2n3 convergence for ChiUUUUUURegQbz = on case.  
 fig002: n1n2n3 convergence for ChiUUUUUURegQbz = off case.  
 (ChiUUUUUURegQbz in explained in General section in this manual).  
  
 As you see, k points convergence looks a little better in ChiUUUUUURegQbz=off  
 (mesh not including gamma). However a little ploblem is that its thereshold around   
 0.5eV is too high and slowly changing.  
  
 fig003: Alouanis'(from Arnaud) vs. ``ChiUUUUUURegQbz = on'' vs. ``ChiUUUUUURegQbz = off''  
 As you see, the threshold of the Red line (20x20x20 ChiUUUUUURegQbz=on) and Alouani's   
 are almost the same, but the red line is too oscilating at the low energy part.  
 On the other hand, ``ChiUUUUUURegQbz = off'' in Green broken line is not so satisfactory  
 at the low energy part.   
  
 fig.gasUUUUUUepsUUUUUUkconf.pdf shows the convergence behevior of epsilon for   
   
   
2.$q \to 0$ convergence (this is related to whether ChiUUUUUURegQbz=on or off).  
 If you use very small q like q=0.001 is GaAs, it can cause a problem.  
 Use q=0.01 or larger (maybe q=0.02 or more is safer).   
 Very small q can give numerical error for high-energy region.  
  
 In fig004, we show the high energy tail part of Im $\epsilon(\omega)$ for GaAs case.  
 At q=0.01 (this means q= 2\*pi/alat \* (0 0 0.01)), the imaginary part  
 is a little too large . Less than 80eV, q=0.02 gives good results when compared with  
 other high q results, though it still has noise above 80eV.  
 In fig005, I showed the same results compared with Alouani's (his is up to 40eV).  
 Both gives rather good agreements. As you see, q=0.06 or above might be necessary  
 to get reasonable convergence for high energy part abouve 40eV.  
  
 We have to be careful for this poorness in high energy part--- it may effect  
 low-energy Re[$\epsion$] through KK relation. However this can be very small  
 ehough.  
 In fig.gasUUUUUUepsUUUUUUqconv.jpg, we checked the convergence of eps (\omega=0,q) for q \to 0.  
 As you see, it gives convergence, however, q=0.01 is a little out of   
 curve---this should be because of the poorness in the high energy part.  
 so q=0.02 or q=0.03 is safer, and you can get eps within 1 percenr accuracy.  
  
3. Including Core for dielectric constant is dangerous.   
 It can cause very poor results if you include core part in GWinput.  
 You need to include core just as valence (with local orbital).  
  
 In fig008, we showed core effects. It starts from \approx 16eV   
 (this is core to conduction transition).  
 fig007 showd the check about the q point dependence---even with large q,  
 it would not change.  
 These shows that the core excitation can have larger energy range.  
 This is in contrast to the valence case   
 (then the most of excitaion is limited to less than 10eV).  
 We have to be careful for such high-energy exciation... The LMTO basis might  
 be not so good for high energy.  
  
4. basis set.  
 Use QpGcutUUUUUUpsi \approx 3.0 a.u. or so (as same as GW calculation).  
 In the case of epsPP\* mode,   
 QpGcutUUUUUUcou can be very small--- In our codes now,   
 ngc>=1 should be for all q vector shown in lqg4gw02 (output of echo 2|qg4gw).  
 [In principle, it should be only for the q vector for which we calculate epsilon.  
 But there is a technical poorness in our code---  
 (maybe) a problem here; the plane-wave part of the eigenfunction generated   
 in lmfgw is not correctly passed to lmf2gw when ngc=0].  
  
  
-- epsUUUUUUlmfh: including LFC ----------------------------------  
To include eps with LFC, do epsUUUUUUlmfh.   
But lcutmx=2 seems to be good enough to get 0.5 percent error (maybe better than this).  
Test it 10x10x10 or so. (I need to repeat if necessary).  
Further you can use smaller QpGcutUUUUUUcou like 2.2 or so,   
with rather smaller product basis (up to p timed d, not including f).  
  
Note: epsPPUUUUUUlmfh is designed to use good basis to calculate eps   
without LFC. This is usually in agreement with what you obtained by epsUUUUUUlmfh;  
however it can give slight difference when you use small product basis.  
  
  
---Summary --------------------  
So in conclusion, I think a best way to do is  
  
1. set q=0.02 [q=2pi/alat(0 0 0.02)] or so for GaAs case.  
 If you want to check, do q=0.03 and q=0.06 also.  
  
 ``ChiUUUUUURegQbz = off'' is better for matrials like GaAs with direct gap.  
  
2. You can use small QpGcutUUUUUUcou but all ngc should be one or more.  
  
3. As for the Product basis setting in epsPP\* scripts, only  
 lcutmx and tolerance (this can be like 0.001 or so) are relevant.  
 E.g. set lcutmx=4 or so.  
  
4. Do nk=20 18 16 and take interpolarion to determine eps(omega=0, q=0).  
  
5. To get eps with LFC, set QpGcutUUUUUUcut as xxx, and set lcutmx=2 where  
 (occupied sp) \timex (unoccupied spd) are included.  
 But correct EPS\*.nolfc.d is rather from epsPPUUUUUUlmfh script.

fig001

fig002

fig003

fig004

fig005

fig007

fig008

# Utility

1. **structure tool** viewvesta,ctrl2vasp,vasp2ctrl
2. **Other calculation examples** A recent paper by Deguch et al contains a link to a sample package.
3. **cif to PROCAR**

* ./cif2cell --vasp-cartesian --vasp-format=5 cifs/BaTiO3UUUUUUcubic.cif

1. In cases to treat magnetic systems which have intrinsic magnetic fluctuations, we may need to be careful about initial condition or mixing procedure to get convergence. In cases, we need to start from LDA+U results as initial condition from which we start QSGW. Let me know about such trouble. [↑](#footnote-ref-23)
2. In the case of using sigm.\*, total energy shown in the console output (also in save.\*) are just the indicator, not the meaningful total energy [↑](#footnote-ref-24)
3. current implementation have not yet efficiently use this locality; this must allow us to speed up one-body problem solver. [↑](#footnote-ref-26)
4. Exactly speaking, we use LDA idea for efficient implementation of QSGW; thus obtained results are slightly dependent on the choice of LDA or GGA [↑](#footnote-ref-27)
5. We undo electron-phonon effect (0.06eV) and spin-orbit effect (0.11eV) from the true the experimental value 1.52 eV. [↑](#footnote-ref-28)
6. exactly speaking, we have range cutoff for Fock exchange term in the HSE functional in addition [↑](#footnote-ref-30)
7. If you use (Hartree-Fock limit), the band gap of Si becomes 20eV or something. [↑](#footnote-ref-31)
8. people often pronounce this “H-naught” [↑](#footnote-ref-32)
9. ctrlgenM1.py exists originally at ecalj/lm7K/ (ctrlgenM1.py was already copied to your BINDIR= defined in ecalj/Install.ifort in the installation). [↑](#footnote-ref-42)
10. For these variables, we can overlay values when we start programs. For example,’lmf -vdalat=0.1 si’ means that alat is recorded in save.si file. [↑](#footnote-ref-44)
11. This may looks slightly uncomfortable since the end of range of ATOM is not clearly shown; it end just at the next ATOM token or new category. [↑](#footnote-ref-45)
12. Correlation part due to cores is neglected. In future, we will switch to a version without PB for core part to reduce computational time and for numerical accuracy. [↑](#footnote-ref-78)
13. Note that QPU also implies QPD and so on. U is for up D is for down spins. [↑](#footnote-ref-97)