

London Dispersion Corrections to Density Functional Theory for Transition Metals Based on Fitting to Experimental Temperature-Programmed Desorption of Benzene Monolayers

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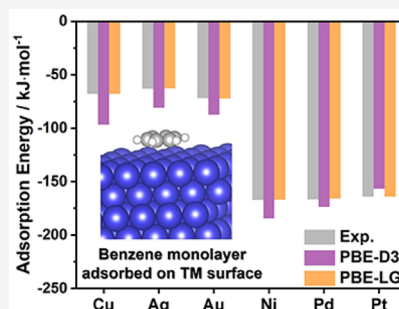
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ABSTRACT: Standard implementations of generalized gradient approximation (GGA)-based density functional theory (DFT) describe well strongly bound molecules and solids but fail to describe long-range London dispersion or van der Waals (vdW) attraction interactions that are important in molecular crystals and two-dimensional solids. To provide accurate values for the vdW distance and energies for the metals Cu, Ag, Au, Ni, Pd, and Pt, we determined empirical vdW corrections to Perdew, Burke, and Ernzerhof (PBE) DFT by fitting the experimental adsorption enthalpies measured by temperature-programmed desorption (TPD) from benzene monolayers by Campbell and co-workers (*J. Phys. Chem. C* 2016, 120, 25161–25172). Benzene physisorbed to these metals without chemical reaction; therefore, we consider the bonding to be vdW. We use the low gradient form for the vdW corrections, $E_{\text{vdW-LG}} = -C_{6\text{LG}}/[R^6 + R_{\text{vdW-LG}}^6]$ with just two parameters per atom ($C_{6\text{LG}}$ and $R_{\text{vdW-LG}}$). This LG form leads to negligible changes in bond distances and angles, so adjusting the parameters should not sacrifice accuracy for the bonding interactions. We demonstrate that the parameters fitted to benzene also describe well the physisorption enthalpies for other hydrocarbons (naphthalene, cyclohexane, methane, ethane, and propane) on Pt. We also report low gradient vdW correction parameters for the noble gases that fit the equilibrium lattice parameter and heat of vaporization of the crystals.



It is well-known that GGA-based DFT functionals, such as PBE developed by Perdew, Burke, and Ernzerhof, fail to

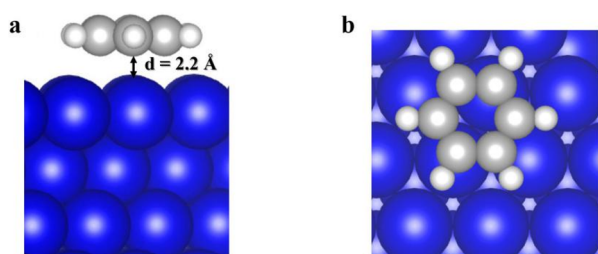


Figure 1. Optimized physisorbed structure of benzene (at the coverage of 0.111 ML) on the Pt(111) surface. (a) Side view. The minimized structure has the benzene plane midpoint 2.20 Å above the midplane of the top Pt layer. (b) Top view. The carbon atoms are in bridging positions, while the H atoms are nearly on-top. The colors are Pt in blue, C in gray, and H in white.

describe the long-range van der Waals (vdW) attraction interactions (London dispersion). This is because vdW attraction arises from instantaneous electron correlations between electrons on adjacent atoms, which is not captured by the average electron density.² More rigorous functionals with London dispersion included are under development,^{3–5} but they are not yet ready for transition metals. As we know,

Grimme's DFT-D methods are quite popular and have been widely validated in many molecular and periodic systems. For instance, for nonmetallic systems that form molecular crystals, a practical solution was developed by Grimme with parameters optimized by Becke–Johnson.^{6,7} Indeed, for nonmetallic systems involving first- and second-row elements (the C and Si rows), this D3 correction does quite well in reproducing the experimental equation of state of molecular crystals up to ~100 GPa (after including the phonons to calculate zero-point energy and temperature corrections to the enthalpy).⁸

However, metallic systems do not exhibit molecular crystals, making it difficult to estimate accurate vdW corrections. The target of this Letter is to accurately describe the organic–metal interactions important for electrochemical reactions, such as hydrogen evolution reaction (Pt),⁹ oxygen reduction reaction (Pt),¹⁰ and CO₂ reduction (Au, Cu, and Ag),^{11,12} when 0.1 eV energy difference makes large deviation in prediction¹³ as measured using temperature-programmed desorption (TPD) by Campbell and co-workers,^{1,14} who measured the accurate

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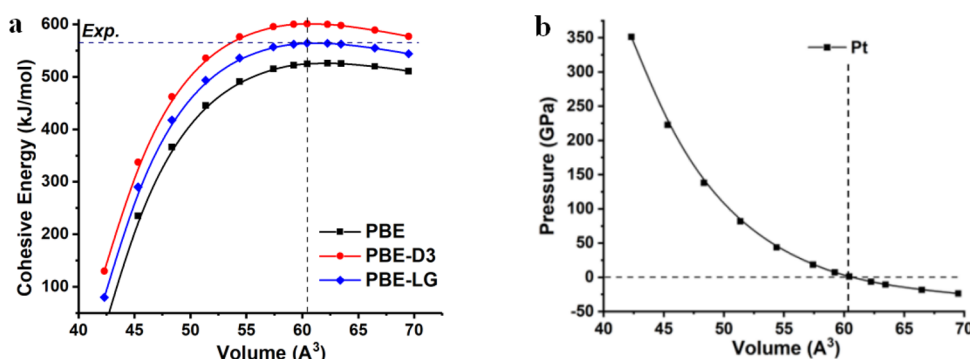


Figure 2. (a) Comparison of the calculated cohesive energy of Pt crystal for various DFT methods compared to the Campbell experiments. (b) Predicted and experimental equation of state for Pt crystal at 0 K. Here we used the minimized structure from the DFT calculations (with or without vdW corrections) and the calculated vibrational energies to obtain the energy and pressure at the experimental temperature.

Table 1. Adsorption Energies from DFT Calculations to a Single Layer of Physisorbed Benzene Including vdW Corrections Compared to Experiment^a

	adsorption of benzene on metal surface (kJ·mol ⁻¹)			R_{vdwLG} (Å)	$C_{6\text{LG}}$ (eV)	S (cross-term factor for organic)
	$E_{\text{PBE-D3}}$	$E_{\text{PBE-LG}}$	exptl ^{1,9}			
Pt (3 × 3)	-156.60	-164.02	-164 (T = 300 K)	6.492	2520.5	0.5919
Cu (3 × 3)	-96.72	-67.98	-68 (T = 225 K)	4.594	455	3.2500
Ag (3 × 3)	-80.72	-62.66	-63 (T = 210 K)	4.992	410	0.2680
Au (3 × 3)	-87.24	-72.12	-72 (T = 230 K)	6.898	12257	0.1872
Ni (3 × 3)	-184.39	-166.90	-167 (T = 90 K)	4.492	99.49	-0.3462
Pd (3 × 3)	-173.70	-165.67	-166.5 (T = 300K) ^b	4.996	356	-0.2089

^aIn this table, the metal vdW parameters are for M–M, but we need to scale C_6 by the factor S before applying the combination rule for Pt–C and Pt–H. Here, we combined the minimized structure from the DFT calculations (with vdW corrections) with calculated vibrational energies to obtain the binding enthalpy at the experimental temperature. Then we determined the two-body R_{vdw} and C_6 parameters to reproduce the experimental binding enthalpy. ^bThe binding of benzene to Pd used the average for the data of Ni and Pt because no experimental data is available (suggested by Charles T. Campbell, private communication).

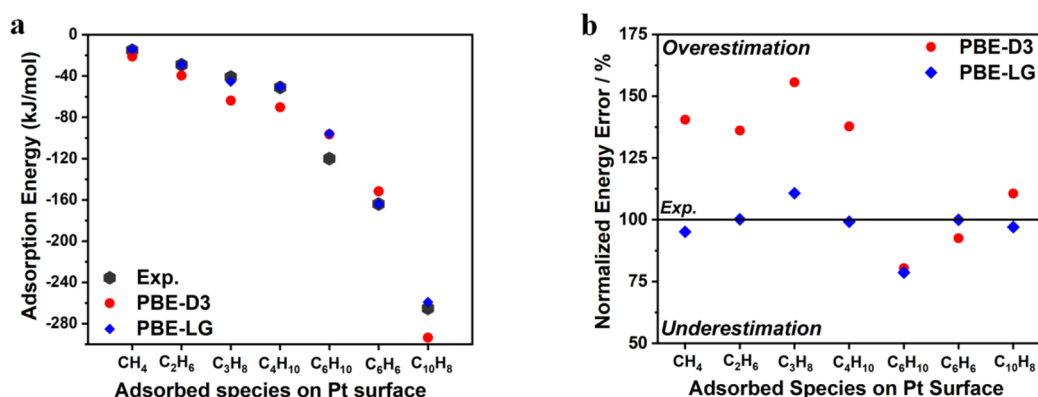


Figure 3. PBE-LG predicted adsorption energy for 6 hydrocarbons and benzene (training set) on Pt(111) surface. (a) Comparisons of adsorption energies with experiment and (b) normalized relative error for each method. Here we have used the minimized structure from the DFT calculations (with or without vdW corrections) and the calculated vibrational energies to obtain the energy and pressure at the experimental temperature of 300 K.

physisorption energetics for the first monolayer of benzene on these metals.

We also determined the vdW parameters for the noble gases based on fitting the crystal structure, heat of vaporization, and equation of state.

We will focus on two sets of atoms in this work:

- Transition metals of columns 10 and 11 on the periodic table: Ni, Pd, Pt, Cu, Ag, and Au, where we fit the LG form

$$E_{ik\text{-dispersion}} = - \sum_{ik,i < k}^N \frac{C_{6\text{LG-ik}}}{R_{0ik}^6 + R_{\text{vdwLG-ik}}^6} \quad (1)$$

to the experimental temperature-programmed desorption (TPD) data of physisorbed benzene.

Here $C_{6\text{LG-ik}}$ and $R_{\text{vdwLG-ik}}$ are optimized to fit binding of physisorbed molecules for atoms i and k and R_{ik} is the variable distance between atoms i and k .

- The noble gas crystals (He, Ne, Ar, Kr, Xe, and Rn), where we fit the LG van der Waals corrections to the experimental lattice parameters, the heat of vaporization, and equation of state.

Physisorbed Enthalpies for Columns 10 and 11 of the Periodic Table. We will discuss the Pt case in detail followed by a summary of results for the other metals.

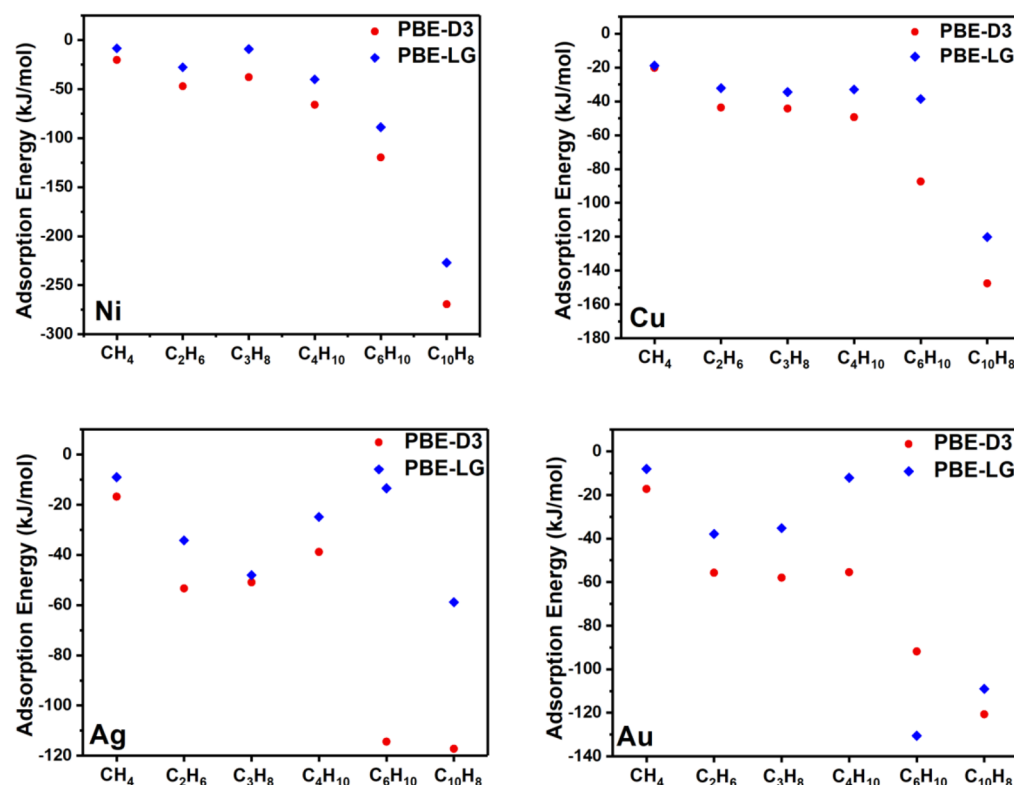


Figure 4. Predicted physisorption enthalpies of hydrocarbons on metals (Ni, Cu, Ag, and Au) at the same temperature as in Table 1.

Table 2. Cohesive Energy from Various DFT Methods Compared to Experiment (Extrapolated to 0 K)^a

	cohesive energy (kJ·mol ⁻¹)			R_{vdwLG} (Å)	$C_{6\text{LG}}$ (eV)
	$E_{\text{PBE-D3}}$	$E_{\text{PBE-LG}}$	exptl ¹⁵		
Pt	602.3	566.2	563.5	6.492	4258
Cu	384.2	337.3	336.7	4.594	140
Ag	290.6	285.0	284.6	4.992	1530
Au	342.0	367.8	367.6	6.898	12257
Ni	497.7	428.4	428.4	4.492	-287.41
Pd	501.1	375.3	375.3	4.996	-1704

^aHere we used the minimized structure from the DFT calculations (with vdW corrections) and the calculated vibrational energies to obtain the energy and pressure at the experimental temperature.

Physisorbed Benzene. The TPD experiments find that the first layer of physisorbed benzene packs with one molecule per 3×3 cell on Pt(111), leading to 0.111 ML coverage (Figure S1).^{1,14} Experiments find that desorption of this layer at 300 K requires 164 kJ/mol.

PBE-D3 leads to the optimized geometry shown in Figure 1, which has the benzene plane midpoint 2.21 Å above the midplane of the top Pt layer, with a DFT binding energy of -156.6 kJ/mol.

To compare with the free energy as measured experimentally, we calculate the phonons of the benzene layer and the top layer of Pt and correct for the zero-point energy (ZPE) and the specific heat to obtain the enthalpy at 300 K (Figure 2). We also calculate the interaction energy between the benzene (within this layer) and Pt to obtain the energy to desorb benzene molecules from the surface (see more details in Table S2).

We want to modify the vdW interactions to fit the experimental binding enthalpy, but we assume that the equilibrium distance of the benzene plane to the Pt surface of

2.21 Å from PBE-D3 may be accurate. Thus, we fitted the well depth to adjust the binding energy, while keeping the distance unchanged. Then, using our new parameters we make any further adjustments needed. The new parameters, $C_{6\text{LG-PtPt}}$ and $R_{\text{vdwLG-PtPt}}$ are listed in Table 1. Our new binding energy of benzene on Pt of -164.02 kJ/mol reproduces the experimental results of -164 kJ/mol at 300 K. This changes the Pt surface benzene physisorption distance slightly to 2.20 Å.

Predicted Physisorption of Six Other Organics to Pt(111). Having optimized $C_{6\text{LG}}$ and R_{vdwLG} for benzene, we now predict the binding energy for 6 other organics measured by Campbell. Figure 3 shows that this leads to a reasonable agreement with the experiment, indicating that PBE-LG can be used for predicting binding for other physisorbed molecules.

We also predicted the adsorption behavior of these six hydrocarbons for Cu, Ag, Au, Pd, and Ni, with results as in Figure 4 and Table S3.

Bulk Cohesive Energies. We had hoped that the LG parameters for benzene physisorbed on Pt might also lead to the correct cohesive energy of Pt. Thus, we calculated the cohesive energy of pure bulk Pt (fcc) using the LG parameters (Table 2). Using the same R_{vdwLG} as for binding benzene to the surface we find that $C_{6\text{LG-PtPt}} = 4258$ eV is required for the bulk cohesive energy, whereas $C_{6\text{LG-PtPt}} = 2520.5$ eV was required for the physisorbed benzene, which is reduced by a factor of 0.5919. This scaling factor is different for each metal element. Thus, we cannot use the same C_6 for bulk Pt as for physisorption on Pt surfaces.

Cu, Ag, and Au. Figure 5 shows the calculated cohesive energy of pure fcc Cu, Ag, and Au crystal using optimized LG parameters, which are in good agreement with the experiment.

Ni and Pd Cases. The optimal LG parameters for Ni and Pd are summarized in Table 1. As discussed in the Supporting Information, the calculation of Ni atom in VASP assumes that each of the five 3d orbitals has 8/5 electrons, which leads to a

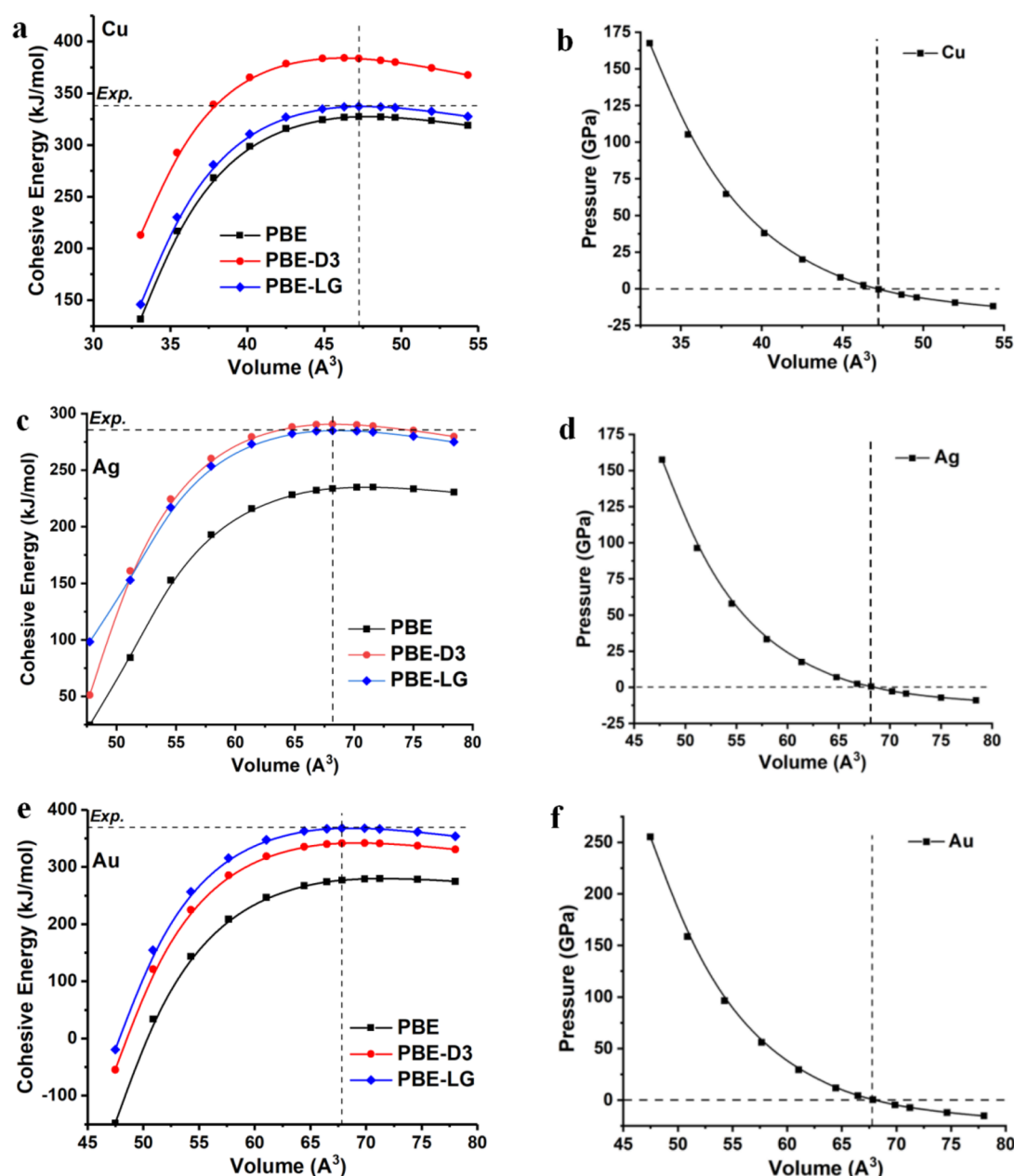


Figure 5. Calculated cohesive energy for crystals using PBE, PBE-D3(B.J.), and PBE-LG corrections. Also, the equilibrium volume for (a and b) Cu, (c and d) Ag, and (e and f) Au. Here, we used the minimized structure from the DFT calculations (with or without vdW corrections) and the calculated vibrational energies to obtain the energy and pressure at the experimental temperature.

mixture of the 3F and 3P states for the $(4s)^2(3d)^8$ configuration for Ni. This leads to a Ni atom energy too high by $0.535 \text{ eV} = 51.65 \text{ kJ/mol}$. We corrected this in the figure and tables (see Figure 6). To match the experimental cohesive energy, we used a negative C_6 to lower the overestimation from pure PBE.

PBE fails to provide consistent predictions of the cohesive energies of metals. In the metals investigated in this work, PBE underestimates the cohesive energies for metals like Cu, Ag, and Au, but it overestimates the cohesive energy of Ni.¹⁶ This inconsistency of PBE poses a great challenge to any postage correction with pure attraction. Advanced methods are expected to solve the above problem. The advantage of the effective method with embedding multiple effects into parameters is the minimal cost added to reproduce the known experimental

results, which are important for applications, such as electrochemical reaction, that require accuracy within 0.1 eV .

van der Waals density functional, such as strongly constrained and appropriately normed (SCAN) with the nonlocal correlation part from the rVV10, has been demonstrated to be reliable in describing the interaction between organic and metals. Although the computational cost is about 10 times more than PBE-LG, SCAN+rvv10 provides reliable predictions when experimental results are not available. The advantage of PBE-LG lies in its high efficiency and accuracy for systems within (or close to) the training set.

Noble Gas Crystals. The standard D3 parameters for the noble gas atoms match accurately the two-body interactions for the diatomic system, including $1/R^8$ terms and $1/R^6$ terms as in eq 1. However, this overestimates the binding in the crystal. The

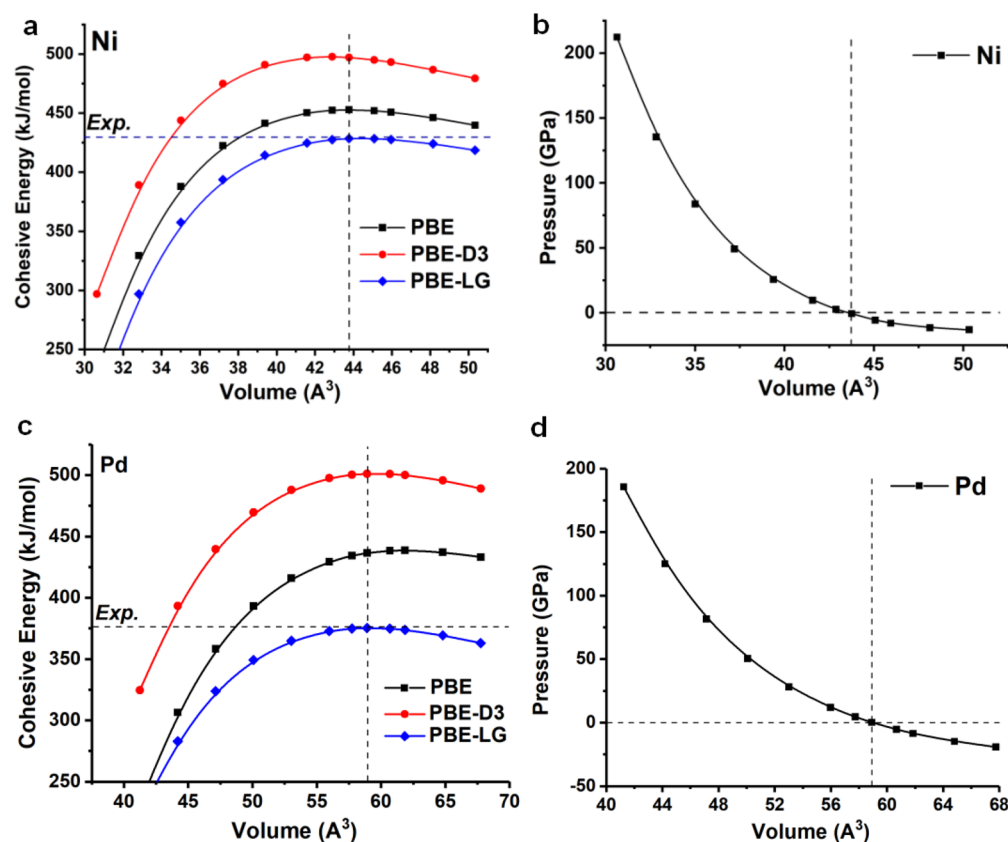


Figure 6. Comparison of cohesive energy of (a and b) Ni and (c and d) Pd between pure PBE, PBE-D3(B.J.), and PBE with LG corrections.

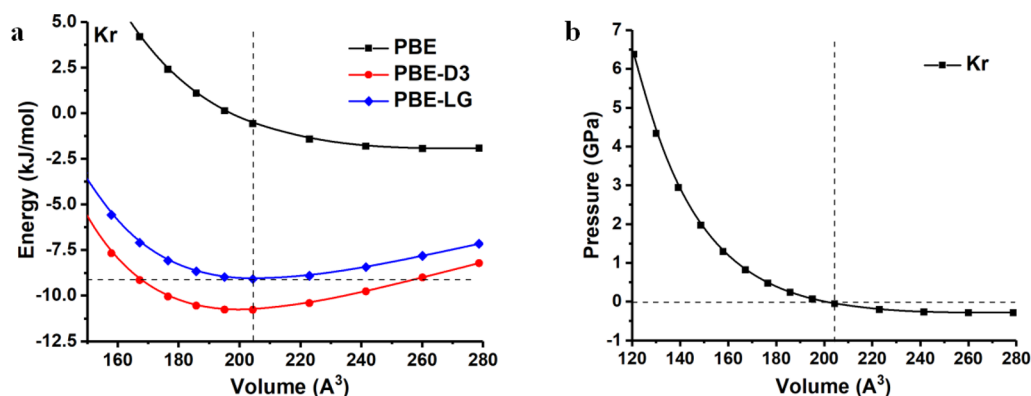


Figure 7. (a) Comparison of the calculated heat of vaporization of Kr crystal for various DFT methods compared to experiment. (b) Predicted and experimental equation of state for Kr crystal at 26 K.

problem is that the third-order Axelrod–Teller dispersion interactions are also important in the crystal, which tends to reduce the net cohesive energy. We expect that such higher-order dispersion probably plays a similar role in all condensed systems, even if not crystalline, but including them in the simulations would be impractical.

Instead, we fit the two-body LG van der Waals to the experimental lattice parameters, the heat of vaporization, and equation of state. These two-body terms implicitly include the three-body Axelrod–Teller terms.

We will discuss the case of Kr in detail and summarize the results for the other noble gas crystals (Figure 7). Table 3 shows the PBE-D3 equation of state (EoS) for Kr and the EoS after correcting for zero-point energy (ZPE) and the temperature dependence of the free energy to 26 K, the temperature for the

experimental EoS. The predicted heat of vaporization at 26 K, ΔH_{vapor} , is compared with the experiment in Table 3, where we see that it is 120% too large.

The EoS for the other noble gases is included in the Supporting Information. The comparison of the ΔH_{vapor} for the other noble gases is in Table 3.

Trends in vdW Parameters. We compare C_6 and R_0 for selected transition metal and corresponding Noble gases in Figure 8. It appears that the slope of $\log C_6$ vs R_{vdw} is steepest for Ne–Ar, which do not have core d electrons, less steep for Ni–Cu–Kr and Pd–Ag–Xe which have d core electrons but not f, and even less steep for Pt–Au–Rn which also has f core electrons; perhaps these trends might be used to estimate corrections for other metals.

Table 3. Heat of Vaporization of Rare Gas Face-Centered-Cubic Crystal Systems (He, Ne, Ar, Kr, Xe, and Rn) Calculated Using PBE-LG Method, Compared with PBE-D3 (B.J.) Correction and Experiments^a

	heat of vaporization (kJ·mol ⁻¹)			R_{vdwLG} (Å)	$C_{6\text{LG}}$ (eV)
	$E_{\text{PBE-D3}}$	$E_{\text{PBE-LG}}$	exptl		
He	0.63	0.09	0.08 ¹⁷ (T = 0 K)	2.2239	0.7916
Ne	2.40	1.91	1.92 ¹⁸ (T = 24 K)	3.0534	10.2421
Ar	7.16	7.66	7.73 ¹⁹ (T = 24 K)	3.6418	236.6570
Kr	10.78	9.08	9.02 ¹⁹ (T = 26 K)	4.7630	815.7240
Xe	19.64	14.93	14.94 ¹⁹ (T = 24 K)	4.9990	2434.3320
Rn	18.22	20.00	19.98 ¹⁸ (T = 24 K)	6.8990	14001.5740

^aHere we have used the minimized structure from the DFT calculations (with vdW corrections), calculated vibrational energies, and used these to obtain the energy and pressure at the experimental temperatures.

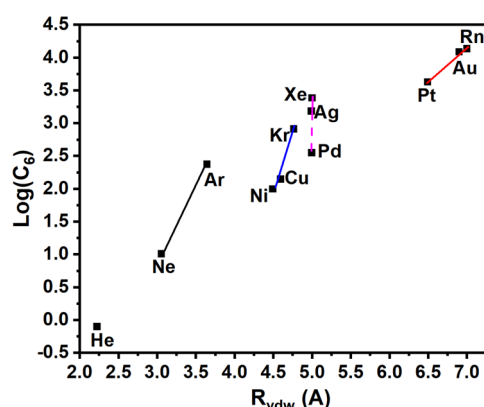


Figure 8. Plot of R_{vdw} versus $\log C_6$. The colored lines are linear fits for elements with similar core electron distributions. (The point here for Pd is only from predictions, along with the extended dashed line.)

The current strategy can be extended to heteroatom-containing organic molecules, such as methanol. The adsorption energy of methanol on Pt predicted from PBE-LG is -0.56 eV, which is almost the same as the experiment of -0.59 eV. We will extend our current scheme to cover more metal–organic systems when experimental results are available.

In summary, we developed new vdW correction parameters for group 10 and 11 transition metals by fitting to experimental binding cohesive energy and for all six noble gases (i.e., He, Ne, Ar, Kr, Xe, and Rn) by fitting the heat of vaporization. This led to new parameters, C_6 and R_{vdw} , derived for the simple low-gradient form. Using the newly fitted LG parameters of metals combined with previous LG parameters for C and H, we find a good match to the experimental binding enthalpy for benzene and other hydrocarbons to Pt (111) surface. Also, we predict the adsorption energies of hydrocarbons on other metals using the same methodology. This simple LG correction should provide a good estimate for long-range London dispersion and hence for adsorption behaviors on surfaces.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpclett.0c03126>.

Details of the computational methods; estimation of reaction enthalpy of benzene and other adsorbed species on the metal surface (PDF)

Details of the fitting of the new LG parameters (txt)
Implementation of LG input for VASP (txt)

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Notes

The authors declare no competing financial interest.

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