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Bioinspired Activation of N₂ on Asymmetrical Coordinated Fe Grafted 1T MoS₂ at Room Temperature[†]

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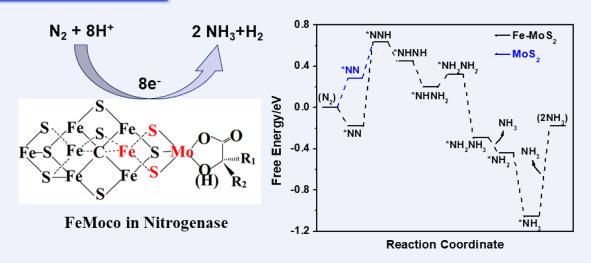
Keywords

Monolayers | Redox chemistry | Electron transfer | Nitrogen fixation | 1T MoS₂

Main observation and conclusion

Inspired by the nitrogen fixation process on MoFe nitrogenase, asymmetrical coordinated Fe grafted onto 1T MoS₂ were successfully synthesized. The unique electron-rich structure with asymmetrical coordination made the 1T Fe_{0.1}Mo_{0.9}S₂ layered material actively react with water and dinitrogen at room temperature and atmosphere pressure. Subsequently, ammonia can be produced with a yield of $\sim 800~\mu mol~(NH_4^+)~g^{-1}$ (12.5% yield in mole). The activation, fixation and reduction of dinitrogen were confirmed by isotopically labeled experiments. The location and the specific coordination environment of grafted Fe in Fe-Mo-S were further determined by X-ray absorption spectroscopy analysis. Our work demonstrates that the nitrogen fixation and reduction for ammonia at room temperature without any chemical and electrochemical assistance is distinctly different from traditional bionic-inspired nitrogen fixation process. The mechanism of the activation and reduction of N2 was further investigated by density functional theory calculation and Raman spectra. Compared with 1T MoS2, the enriched electron nature and asymmetrical coordination of Fe in Fe-Mo-S materials play a critical role in the bioinspired activation of N_2 at ambient condition.

Comprehensive Graphic Content



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Supporting Information



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Dedicated to Professor Yitai Qian on the Occasion of His 80th Birthday.

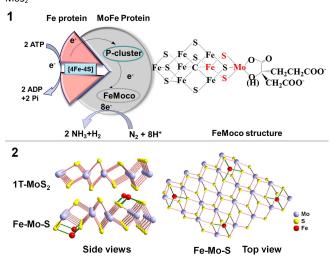
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Background and Originality Content

Nitrogen fixation is an extremely vital process that nourishes the life on earth. $^{[1-2]}$ Since the development of Haber-Bosch process (HBP) in 1900s, synthesis of ammonia at industrial scale has utilized the dinitrogen (N2) that is rich in atmosphere (78% of the air). It significantly met the demand to produce chemical fertilizers, in turn supporting the population growth from 1.8 billion to 6.5 billion during the last hundred years. [3] However, the conversion of N₂ to ammonia requires harsh conditions like high pressure and temperature (~300 atm and 400-600 °C), [4] as the bond disassociation energy for N≡N is 945 kJ/mol. [5] In addition, the purification of nitrogen and hydrogen also consumes lots of energy, which, consequently raises 2% of global energy consumption and 2%—3% of global CO₂ emission. [1-2,6] Therefore, exploring N₂ activation and reduction strategies at ambient conditions (i.e., room temperature and atmospheric pressure) is crucial to reduce the energy consumption and improve the efficiency of ammonia production.

In nature, enzymes accomplished more than 50% of dinitrogen fixation and conversion at room temperature and atmospheric pressure. [7] In the biological processes (Scheme A1), electron transfer is associated when N_2 is activated and inert triple bonds are broken. This is also true in Haber-Bosch process (HBP) as high performance catalysts are used to facilitate the electron transfer. [8-9] Among various reaction steps, the activation of N₂ is found to be the rate-determining step (RDS) in overall ammonia synthesis. However, recent works have shown that the RDS could be shifted from the dinitrogen dissociating activation to the adsorbed N specie's hydrogenation by utilizing electronic donors as substrates for Fe or Rh catalysts, especially electrides with low work functions, such as Li-H, [10] 12CaO·7Al₂O₃ (C12A7:e $^-$), [11-14] [Y₅Si₃] $^{0.79+}$:0.79e $^-$ [15-16] Detailed experiments and first-principle calculations revealed that negatively charged cobalt atoms in LaCoSi (400 °C, 0.1 MPa) can mediate the electrons from La to dinitrogen, dramatically lowering the activation barrier. Therefore, the active center with enriched electrons and low work function state is suggested to activate N₂ and facilitate the nitrogen fixation process.

Scheme 1 (A1) Schematic illustration of N_2 reduction process on nitrogenase; (A2) the structure of 1T MoS_2 and simulated Fe grafted onto 1T MoS_2



In nitrogenase, researchers have found that the coordination structure of Fe $_7$ MoS $_9$ C cluster plays a critical role in the activation of N $_2$. [7,17-21] Specially, Fe and Mo ions in asymmetrical Fe $_7$ MoS $_9$ C

cluster are in near-tetrahedral and octahedral structure of hexahedral coordination geometry, respectively. [17,21-22] Various works to mimic nitrogenase also suggest the construction of asymmetrical coordinated metal atoms can realize higher NH3 produc- $\mathsf{tion}.^{\text{[23-24]}}$ For instances, Mo catalysts can reach the highest turnover number (TON) of 4350 equiv/Mo by utilizing a strong reducing reagent, Sml₂, as electrons donor. Previously, freshly prepared FeS can react with N_2 and water to form ammonia with a yield of 0.1% per mole FeS. [26] The FeMoS-chalcogels built by symmetrical $[Mo_2Fe_6S_8]^{3+}$ and $[Sn_2S_6]^{4-}$ clusters show limited photocatalytic behavior in nitrogen fixation with a yield of ppm level, which is believed to result from its amorphous nature and symmetrical coordination of Fe and Mo. [27-28] Recently, density functional theory (DFT) calculations predicted that depositing Fe atoms on 2H-MoS₂ to form asymmetrical coordination of Fe, mimicking FeMo-co can spontaneously capture N2 and show electrocatalytic behavior in reduction of N₂ at mild conditions. [29] Hence, combining both functional and structural bionic concepts, molecular engineering of materials with strong electron-donating properties and asymmetric coordinated active center to mimic-FeMo-co would be a promising way to achieve the N2 activation at mild conditions.

Compared with 2H-MoS₂, [30-34] 1T-phase MoS₂ has enriched electron nature (low work function), but its thermodynamic instability restricts its broad applications. [35-37,38-40] Our previous work indicated Co and/or other transition metals can be grafted onto the S atoms at in-plane surface of octahedral 1T S-Mo-S layer via Co-S bonds to form an asymmetric coordinated Co active site, which can simultaneously enhance thermodynamic stability and electron density of 1T phase. [41] Based on these, a new class of N₂ fixation materials were constructed by replacing Co with Fe. Herein, asymmetrically coordinated Fe grafted on 1T MoS₂ as displayed in Scheme A2, was synthesized by a bottom-up method. The covalent bond characteristic of Fe—S is believed to improve the electron density and thermodynamic stability of the 1T MoS₂, and it can elevate the phase transition temperature to 450 $^{\circ}$ C. Notably, iron can act as an electron donor to lower work function. Due to the unique structure of enriched electrons nature and asymmetrical coordination, the obtained 1T Fe_{0.1}Mo_{0.9}S₂ layered material can directly react with water and dinitrogen to form ammonia with a yield of $\sim 800 \mu mol (NH_4^+) g^{-1} (12.5\% yield in$ mole, which is nearly 125 times of previous reported value [26]) at room temperature and atmosphere pressure without electricity or light.

Results and Discussion

Bottom-up synthesis of Fe grafted 1T MoS₂

1T MoS $_2$ and Fe-Mo-S materials were synthesized by utilizing Na $_2$ MoO $_4$ ·2H $_2$ O and S as raw materials and ethylene glycol as solvent via a solvothermal method (see Experimental section for details). 2H MoS $_2$ was synthesized for comparison via a hydrothermal process. Raman spectrum (Figure 1a) of the sample obtained from hydrothermal process shows strong and typical out-of-plane vibration A_{1g} (400 cm $^{-1}$) and E^1_{2g} (370 cm $^{-1}$) of trigonal prismatic (D_{3h}) coordination indicating the formation of hexagonal 2H MoS $_2$. [42] While, the Raman spectra of samples obtained with glycol as solvent presented significant changes. The typical J_1 , J_2 , J_3 vibration and in-plane vibration E^1_{1g} (283 cm $^{-1}$) derived from the MoS $_6$ octahedron (O_h) coordination reveal the formation of tetragonal 1T phase.

The X-ray diffraction (XRD) patterns (Figure 1b) of layered Fe-Mo-S, 2H and 1T MoS_2 indicate low crystalline nature of these materials. Hexagonal 2H-MoS $_2$ synthesized with water as solvent

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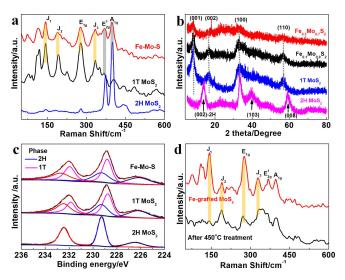


Figure 1 (a) Raman spectra, (b) XRD pattern, (c) XPS of Mo element in 2H, 1T-MoS $_2$ and Fe-Mo-S materials, (d) Raman of Fe-Mo-S materials before and after 450 $^{\circ}$ C treatment.

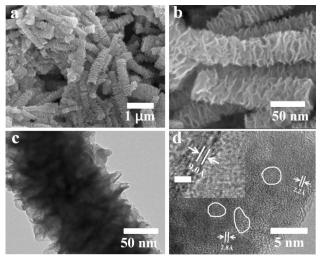


Figure 2 (a, b) SEM, (c) TEM and (d) HRTEM of Fe_{0.1}Mo_{0.9}S₂.

shows well stacked layered structure with a standard (002) plane of 2H-MoS $_2$ (JCPDS card No. 37-1492) at 14.7°, corresponding to an interlayer distance of 6.5 Å. [44-45] While the interlayer distance of samples (1T-MoS $_2$ and Fe-Mo-S) obtained in glycol system was enlarged to 9.0 Å (Figure 2d: Inset), resulting in two related (001) and (002) broad diffraction peaks around 8.1° and 16.8°, respectively. [32] Especially, 1T-MoS $_2$ and Fe-Mo-S materials display one characteristic diffraction peak derived from tetragonal (110) planes around 58°. The decreased intensity of Fe-Mo-S materials reveals the grafted Fe, leading to the distortion of tetragonal 1T structure and developing asymmetrical Fe—S coordination structure. [43] The molar ratio of Fe and Mo detected by inductively coupled plasma atomic emission spectrometry (ICP-AES) was 1:8.6 for Fe $_{0.1}$ Mo $_{0.9}$ S $_2$, which is close to the raw feeding ratio (Table S1).

The X-ray photoelectron spectroscopy (XPS) survey scan of Fe_{0.1}Mo_{0.9}S₂ material indicates the existence of Fe element (Figure S1). High resolution XPS spectra (Figure 1c) of Mo in 2H, 1T-MoS₂ and Fe-Mo-S materials reveal the valence state of Mo⁴⁺ according to 232.2 and 228.8 eV. [40] Meanwhile, compared with 2H-MoS₂, the Mo 3d in 1T-MoS₂ and Fe-Mo-S materials were shifted to low binding energy values revealing the formation of 1T phase with enriched available electrons. Differential scanning calorimetry

(DSC. Figure S2) curve of Fe-Mo-S with only one broad peak around 150—200 °C is derived from the evaporation of H₂O and solvent (interlamellar H2O molecule) indicating the good stability of 1T phase. [39] Raman (Figure 1d) spectrum of Fe-Mo-S after thermal treatment at 450 °C (Ar atmosphere, purity 99.99%) also displays the typical vibration J_1 , J_2 , J_3 of 1T-MoS₂ indicating the enhanced thermodynamic stability of 1T-MoS₂ by grafted Fe. Element mapping analysis of Fe-Mo-S materials presents uniform distribution of Fe atoms into 1T-MoS₂ (Figures S3 & S4). It's also worth noting that SEM and TEM images in Figures 2a, 2b and 2c show the Fe-Mo-S materials are layer-stacked hierarchical structures built by ultrathin nanosheets, which are different from 1T-MoS₂ (Figure S5). Moreover, the lattice fringes with 2.7 Å and 2.2 Å in the HRTEM image (Figure 2d) of $Fe_{0.1}Mo_{0.9}S_2$ plane nanosheets are corresponding to (100) and (110) of tetragonal 1T phase, respectively. [39,43] Some disordered amorphous areas (marked in white circles) can also be observed.

Nitrogen fixation behaviors

The performance of nitrogen fixation on obtained 1T Fe-Mo-S materials was investigated by multiple trails at room temperature and pressure atmosphere. As a result, the formation of ammonia was indeed observed when Fe-Mo-S materials directly reacted with water and dinitrogen at ambient conditions without external light or electricity. To avoid occasional result, the purification of N2 gas and H2O was carried out firstly, and controlled blank experiments were also performed before studying the dinitrogen fixation performance of designed Fe-Mo-S materials. [46-47] Detection of NH₄⁺ ions was also carried out by different methods, including the Nessler and indophenol spectrophotometry, ion chromatography and ¹H NMR to ensure the generation of ammonia during the nitrogen-fixation process. The presence of NH₃ in solution was confirmed by the distinct green color released by the formation of indophenol dye after comparing with blank yellow aqueous solution (Figure 3a-1). [6,28] The Nessler reagent (Figure 3a-2) assay was also conducted, as another colorimetric way, it revealed the presence of ammonia in resulting solution by contrast with a blank (colorless) liquids. [6] For more accurate determination of ammonia production in this reaction, ¹H NMR spectra (Figure 3a) were recorded. The presence of NH₄⁺ ions in the result solution was demonstrated according to the distinct triplet peaks in

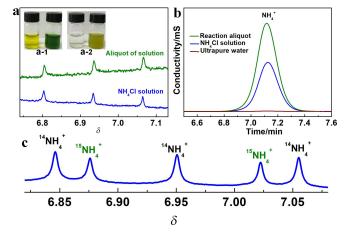


Figure 3 (a) Nuclear magnetic resonance spectra (H) of standard sample (NH₄Cl) and sample to be tested. (Insets, a-1, a-2) Photos of aliquots obtained during nitrogen-fixation process after performing the Nessler and indophenol colorimetric reactions (SI). (b) Representative ion chromatographs of aliquots obtained after nitrogen-fixation showing the presence of [NH₄] $^+$ ions. (c) Nitrogen source isotope analysis: the solution obtained by Fe_{0.1}Mo_{0.9}S₂ reacting with ¹⁵N₂.

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 1 H NMR. Furthermore, ion chromatography is also a direct method for detecting NH₄ $^{+}$ ions (Figure 3b) with a retention time of NH₄ $^{+}$ ions at ~7.2 min. To confirm whether N₂ gas is the source of NH₄ $^{+}$ ions in the current system, isotopic labeling experiments were performed with air and 15 N₂ gas (Figure 3c). 15 N isotope labelling experiment and corresponding 1 H NMR spectra indicate that the source of N element in generated NH₄ $^{+}$ is indeed coming from the N₂ reduction process on Fe-Mo-S. Therefore, it is inferred that the generated NH₄ $^{+}$ ions come from the reaction among Fe-Mo-S material, dinitrogen and water.

To further investigate the behavior of nitrogen fixation on Fe-Mo-S materials, different weights of Fe_{0.1}Mo_{0.9}S₂ were put into water to obtain various dispersions from 0.05 to 0.5 mg Fe_{0.1}Mo_{0.9}S₂ per milliliter water. The results were in good agreement with the cationic chromatographic analysis (Figure 4a) and the color change (Figure 4b) in the illustration. There is an approximately linear relationship between NH₄⁺ ion concentration in the aqueous dispersion and the dosage of Fe-Mo-S materials as displayed in Figure 4c. Also, the yield by each gram of Fe_{0.1}Mo_{0.9}S₂ materials is nearly a constant, yield of ~280 $\mu mol\ g^{-1}$ (4.1% yield in mole, in the initial 1 h). The system gave a fast generation rate of NH₄⁺ in the initial 1 h, while nearly no more NH₄⁺ will be formed after 4 h, and the NH_4^+ yield reached ~800 μ mol (NH_4^+) g⁻¹ (12.5%) yield in mole) (Figure 4d). When air source was changed to the N₂ (99.99%) bubbling condition, the total amount of NH₄⁺ ions increased to ~1000 μmol/g. The reproducibility of the reaction of $Fe_{0.1}Mo_{0.9}S_2$ materials with water under ambient conditions for 6 times (4 h) in parallel (Figure 4e) validated the performance of nitrogen fixation on Fe-Mo-S materials and ruled out this activity as an accidental result. In contrast, 2H-MoS2 presented no ability

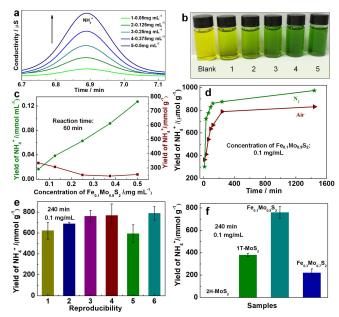


Figure 4 (a) Ion chromatography analysis of ammonia yield with concentration (Supporting Information and Figures S6—S10); (b) Illustration of indophenol colorimetric for NH₃ yield in different concentration of Fe_{0.1}Mo_{0.9}S₂ solutions; (c) Yield analysis of ammonia from two different perspectives (Reaction time = 60 min); (d) Nitrogen fixation performance (reaction time = 1440 min) of Fe_{0.1}Mo_{0.9}S₂ material (concentration = 0.1 mg/mL) under different gas bubbling condition (N₂ and Air); (e) Reproducibility of Fe_{0.1}Mo_{0.9}S₂ reaction with water at atmospheric pressure (reaction time = 240 min, concentration of materials = 0.1 mg/mL); (f) Different nitrogen fixation performances (reaction time = 240 min) of 2H-MoS₂, 1T-MoS₂ and the Fe-Mo-S (molar contents of iron: 10%, 30%) materials (concentration of materials = 0.1 mg/mL).

to convert N₂ to NH₄⁺ in the previously mentioned system. The NH₄⁺ yields of 1T MoS₂ and Fe_{0.3}Mo_{0.7}S₂ were 400 and 300 μ mol/g (6.4% and 4.4% yield in mole), respectively (Figure 4f). It can be inferred that the yield of ammonia has the highest value on Fe-Mo-S materials when the 10% Fe was grafted onto 1T-MoS₂.

It could be clearly understood that only octahedron (O_h) structured 1T-materials have the ability of nitrogen fixation, which may be related to its unique enriched electronic structure and low work function properties, which implies the redox activities of materials. The work function of various materials was measured by UPS spectra. The original Fe_{0.1}Mo_{0.9}S₂ sample shows the smallest work function of 3.5 eV compared with 1T MoS₂ (4.05 eV) and $Fe_{0.3}Mo_{0.7}S_2$ (3.88 eV) (Figures S11 & S12). Low work function indicates the lower binding force of the material surface to electrons, [16] reflecting the layered Fe-Mo-S electron-rich compounds could serve as a favorable electron donors. These could offer available electrons to the lowest unoccupied molecular orbits of the N₂ molecule and active N≡N bonds. After reacting with water in air for 24 h, the work function of Fe_{0.1}Mo_{0.9}S₂ samples increased to 3.9 eV and nearly lost the ability to reduce N₂. These results demonstrate the nitrogen fixation process on Fe-Mo-S materials as a reactive-type process (A + B = C) rather than a catalytic process.

Theoretically, the 1T phase Fe-Mo-S material with enriched electrons feature offers its available electrons to nitrogen and water to form NH_4^+ . EPR spectrum with a sharp signal at g = 2.00(Figure S13) provides fingerprint evidence for the existence of free electrons in $Fe_{0.1}Mo_{0.9}S_2$ (like electrons in n-type semiconductor). The stronger EPR signal of Fe_{0.1}Mo_{0.9}S₂ dispersion (10 min) indicates that the Fe_{0.1}Mo_{0.9}S₂ has more free electrons, which were partially consumed during the process when reacting with water. High resolution XPS spectra (Figures S14-S16) of fresh- $Fe_{0.1}Mo_{0.9}S_2$ and used- $Fe_{0.1}Mo_{0.9}S_2$ samples show the slight shift of Fe 2p signals, indicating the Fe oxidation state is still the mixture of Fe²⁺ and Fe³⁺. While the Mo3d (229.7, 233.1 eV), S2p (162.2, 162.6 eV) peaks shifted significantly to higher binding energy revealing that the available electrons are mainly stored in the 1T S-Mo-S frameworks, which is consistent with previous reports. [39] Besides the free electrons, the change of valance state of elements may also offer redox electrons for the formation of NH₃ and sides reductions of water, such as the generation of H₂ (Figure S26). All these electrons were measured as electron number by a classic iodometric method (Table S2). [12] The electron number of 1 T $Fe_{0.1}Mo_{0.9}S_2$ material changed from 19.4 to 14.9 mmol/g during the process, demonstrating that 4.5 mmol/g electrons were consumed to generate ammonia and other by-products. During this process, 1T Fe_{0.1}Mo_{0.9}S₂ material could utilize 53.3% of these electrons to generate ammonia with a yield of ~800 µmol/g NH₄⁺ (12.5% yield in mole), which is much higher than 1T MoS_2 (38.0%) and $Fe_{0.3}Mo_{0.7}S_2$ (48.9%) and suggests the grafted Fe plays important actions on N₂ activation.

Nitrogen activation mechanism

To get deep insight about actions of Fe on activating N₂, the location and the specific coordination environment of Fe in Fe-Mo-S were further determined by X-ray absorption near-edge structure (XANES) and the extended X-ray absorption fine structure (EXAFS) of Fe_{0.1}Mo_{0.9}S₂ exposed in air. [48-49] XANES spectrum (Figure S17) revealed that the oxidization state of Fe in Fe-Mo-S is a mixture of Fe²⁺ and Fe³⁺. Meanwhile, the comparison of Fourier-transformed EXAFS spectrum (Figure 5a) demonstrated that the first broad peak around 1.7 Å consists of three different bonds: Fe—N, Fe—O and Fe—S. Obviously, no Fe—Fe bond could be found in the obtained materials, showing clear dissimilarity from the spectrum of Fe foil. To further determine the position of Fe in coordinated structure, wavelet transformation and EXAFS fitting were carried out. Wavelet transformation (Figure 5b) showed two

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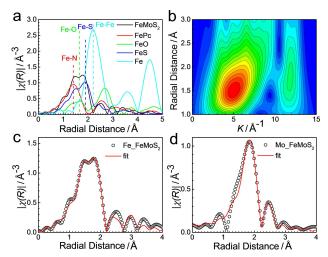


Figure 5 (a) Experimental Fe K-edge EXAFS spectra comparison of $Fe_{0.1}Mo_{0.9}S_2$, FePc (Iron phthalocyanine), FeO, FeS, (b) wavelet transform of experimental Fe K-edge EXAFS spectra of $Fe_{0.1}Mo_{0.9}S_2$, (c) best-fitted Fe K-edge EXAFS spectra of $Fe_{0.1}Mo_{0.9}S_2$ material, and (d) best-fitted Mo K-edge EXAFS spectra of $Fe_{0.1}Mo_{0.9}S_2$ material.

well separated peaks around 5 and 11 Å $^{-1}$ contributed by Fe-N as well as Fe-O, and Fe-S, respectively. While, Fe-N and Fe-O were hardly distinguished by EXAFS. Hence, the model-based EXAFS fitting (Figure 5c) confirmed the mixed Fe-N, Fe-O and Fe-S in the Fe $_{0.1}$ Mo $_{0.9}$ S $_2$, which is in line with the FT-IR (Figure S18) and effectively verifies our assumption that Fe can facilely absorb N $_2$ and activate it in air condition. Meanwhile, the side reaction in present system may exist.

In addition, Mo and Fe EXAFS fitting (Figures 5c & 5d) could also confirm that Fe is grafted onto the surface of MoS_2 instead of substituting the bulk Mo. The fitting results indicated the Fe—Mo bonding length was around 3.12 Å (Figure S19 & Table S3), and the Mo—Mo bonding length was around 2.75 Å (Table S4). Besides, if Fe replaces the bulk Mo atoms, the Fe—Mo bonding length will be closer to Mo—Mo bonding length. Another evidence was that the Fe atoms only have 2 S coordinated with some N and O adsorbed, while the bulk Mo atoms have 6 S coordinated. The wavelet transfer of Mo K-edge EXAFS (Figures S20 & S21, Table S4) also confirmed only Mo—S bonding existing in the first shell, which does not have any N or O adsorbed as surface Fe atoms.

The successful identification of location and specific coordination environment of grafted Fe in Fe-Mo-S offers the opportunity to get deep understanding from theoretical simulation. DFT calculations were carried out to investigate the reaction pathway of N₂ reduction reaction (NRR) on both 1T MoS₂ and Fe grafted 1T MoS₂. Despite the growing interest in N₂RR, the reaction mechanism is still far from clear and in large dispute. In this work, we considered two possible reaction pathways: the associative distal (AD) pathway and the associative alternating (AA) pathway (SI). Previous research indicates that AA pathway is more favorable on both basal plane and edge of MoS₂. [44] With one Fe atom doped, we found that the AA pathway is more favorable, consistent with previous calculations. [44] The free energy profile of AA pathway on Fe doped MoS₂ is shown in Figure 6, while the free energy profile of AD (unfavorable) is shown in Figure S22. As shown in the free energy profile, *NNH formation is the potential determining step (PDS) with the formation energy of 0.81 eV. Compared with pure MoS₂, introducing Fe significantly increases the binding of N₂. Without Fe, MoS₂ shows no tendency to adsorb N₂ as indicated from the positive binding free energy of 0.34 eV, which mostly comes from the decreasing of entropy owing to adsorption. Instead, the presence of Fe significantly increases N₂

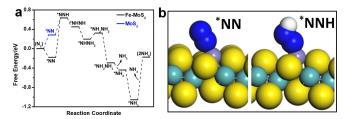


Figure 6 (a) The free energy profile of NRR on the basal plane of MoS_2 and that with one Fe atom doped. (b) The optimized structures of *NN and*NNH of Fe doped MoS_2 from DFT calculations. The colors are cyan for Mo, yellow for S, purple for Fe, blue for N, and white for H.

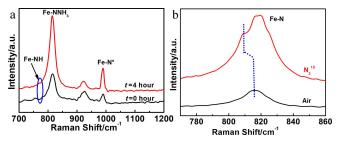


Figure 7 (a) Raman spectra from $750-860 \text{ cm}^{-1}$ (The materials tested were obtained at 0 and 4 h of reacting with water and air, respectively), (b) Raman spectra of $Fe_{0.1}Mo_{0.9}S_2$ material treated with N_2^{15} and air atmosphere.

binding by -0.18 eV. Therefore, DFT calculation results suggest that the presence of Fe significantly increases the NRR performance by increasing N_2 binding, which is the first step towards N_2 reduction product.

We also predicted the vibrational frequencies of important reactive intermediates, such as Fe—N bonds in Fe—N* (1016 cm⁻¹), Fe—*N—NH₃ (815 cm⁻¹) and Fe—*NH (772 cm⁻¹) from DFT calculation, which can be directly compared with the experimental Raman spectra (Figure 7a) showing peaks at 990, 815 and 770 cm⁻¹ (shoulder peak). The consistency between DFT calculations and experiment indicates the existence of possible reactive intermediate after introducing Fe, which demonstrated the important role of Fe site in NRR.

Besides the typical peaks of 1T MoS_2 and Fe-N species, vibration derived from Mo-N was clearly observed (Figure S23). The Mo-N bond is derived from the adsorption of N_2 on the S-Mo-S edge, $^{[39,45]}$ which is usually reported as active sites for NRR process. The abundant S-Mo-S edges may lead to the ability of 1T MoS_2 to active N_2 . The intensity of Fe-N bond is significantly greater than those of the N-Mo bond and other peaks, indicating strong interactions between asymmetrical coordinated Fe and adsorbed N_2 .

Based on theorical predictions, the isotope effect from 14 N to 15 N leads to wavenumber red shift of Fe—N vibrational mode. The activation of N₂ on the Fe-Mo-S was further studied by sealing the fresh material in isotopically labeled N₂ 15 gas for 72 h to perform free absorption. Generally, the formed covalent bonds of Fe—N¹⁴ are hardly to be exchanged with N₂ 15 . While labeled N₂ 15 gas would react with Fe-Mo-S and generate new Fe-N¹⁵ bonds. One can found a new Raman peak of Fe—N¹⁵ at 810 cm⁻¹ (Figure 7b) can be observed besides the Fe—N¹⁴ peak at 815 cm⁻¹. The obvious red shift between Fe—N¹⁵ and Fe—N¹⁴ is consistent with the isotopic effect. Different from covalent bond, the vibration of adsorbed N₂ should show a red shift single peak, which can be clearly observed in the Fourier Transform Infrared spectra (Figure S24). Therefore, it could be deduced that N₂ is easily adsorbed and activated on Fe in Fe-Mo-S material and then converted to

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ammonia.

Based on all our results, the N2 activation and reduction mechanism can be proposed as follows. Fe atoms were grafted onto 1T-MoS₂ via three Fe-S bonds during the solvothermal process based on previous report. [41] The unsaturated coordination of Fe grafted on 1T MoS2 prefers to offer unoccupied 3d orbitals to adsorb and active N₂ to form serial Fe-N bond related species. The electron enriched 1T S-Mo-S framework at lower work function can offer its available electrons to meet with the energy request of N2 reduction process. Driven by available electrons of Fe_{0.1}Mo_{0.9}S₂, the formed intermediate species can be further hydrogenated to produce $\mathrm{NH_4}^+$ once unstable structure meets with missive water. After reacting with water and dinitrogen, the available electron density of materials and the ability of offering available electrons are decreased. Unfortunately, the adsorption free energy of ${\rm O_2}$ on Fe atom site is -0.41 eV, which is (~0.27 eV) stronger than that of N₂ (Figure S25). Therefore, the side oxidation of Fe by O₂ and H₂O in Fe_{0.1}Mo_{0.9}S₂ material is also observed in the Fourier-transform infrared spectroscopy (FT-IR) spectra, and H₂ byproducts can also be detected during the N₂ reduction process (Figure S26).

Conclusions

In summary, inspired by the nitrogen fixation process on MoFe nitrogenase and developing industry of catalysts, asymmetrical coordinated Fe grafted onto 1T MoS₂ was successfully synthesized by a bottom-up method. The introduction of iron increased the electron density of materials and lowered work function rendering these materials as an electron donor. The unique enriched electronic and asymmetrical coordination structure made 1T Fe_{0.1}Mo_{0.9}S₂ layered material to actively react with water and N2. Consequently, the formation of ammonia with a yield of \sim 800 μ mol (NH₄⁺) g⁻¹ Fe_{0.1}Mo_{0.9}S₂ (12.5% yield in mole) at room temperature and atmosphere pressure was accomplished without electricity or light, which is greatly different from traditional bionic-inspired nitrogen fixation process. The fixation of N₂ was carefully investigated and confirmed by various isotopically labeled experiments. Moreover, the N₂ activation mechanism on Fe grafted 1T MoS₂ has also proposed based on the various controlling experiments, structure analysis based on XANES and EXAFS. The mechanism of the activation and reduction of N2 was further investigated by DFT calculation and Raman spectra. It is believed the enriched electron nature of 1T MoS₂ structure and asymmetrical coordination of grafted Fe play key actions on the activation of dinitrogen on Fe-Mo-S materials at ambient condition. This work paves the potential way to develop highly electron utilization ratio nitrogen fixation materials for electro- or photo-NRR process via utilizing sustainable energy.

Experimental

All the experimental details can be found in Supporting Information.

Supporting Information

The supporting information for this article is available on the WWW under https://doi.org/10.1002/cjoc.202000675.

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