

- Phenomenological description of ferromagnets: Ginzburg-Landau theory
- Anisotropic ferromagnetic materials: ferromagnetic domains, hysteresis loops, engineering domain structures
- Origin of domains: energetics of domain walls. Single-domain particles.

# ENTROPY OF A PARTIALLY POLARIZED STATE

$$\begin{cases} N = N_{\uparrow} + N_{\downarrow} \\ \langle s_z \rangle = \frac{1}{2} \frac{N_{\uparrow} - N_{\downarrow}}{N} \end{cases} \Rightarrow \begin{cases} \frac{N_{\uparrow}}{N} = \frac{1}{2} + \langle s_z \rangle \\ \frac{N_{\downarrow}}{N} = \frac{1}{2} - \langle s_z \rangle \end{cases}$$

Entropy (relative to the  $\langle s_z \rangle = 0$  state)

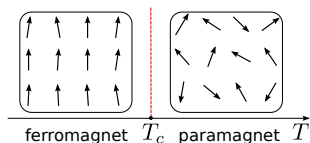
$$S = k_B \ln \mathbb{P} \qquad \mathbb{P} = \frac{1}{2^N} C_N^{N_{\uparrow}} = \frac{1}{2^N} \frac{N!}{N_{\uparrow}! N_{\downarrow}!}$$

If  $\langle s_z \rangle \rightarrow -\langle s_z \rangle$ , then  $N_{\uparrow} \longleftrightarrow N_{\downarrow}$  and  $S = \text{inv.}$

Entropy is thus an even function of  $\langle s_z \rangle$ , and for  $\langle s_z \rangle \rightarrow 0$

$$S = a \langle s_z \rangle^2 + b \langle s_z \rangle^4 + \dots$$

# SPONTANEOUS SYMMETRY BREAKING IN MAGNETIC PHASE TRANSITIONS



Order parameter:

“Paramagnetic” phase:  $\mathbf{M} = 0$

Ferromagnetic phase:  $\mathbf{M} \neq 0$

Thermodynamically, “PM” phase is *more* symmetric than FM phase: having finite  $\mathbf{M}$  breaks *rotational invariance* and *time-inversion symmetry*.

Indeed, as  $t \rightarrow -t$ ,  $\mathbf{v} \rightarrow -\mathbf{v}$ , and ( $\mathbf{M} = n_{\text{at}} \mathbf{m}$ )

$$\mathbf{m} \equiv \frac{e}{2c} \sum_i \mathbf{r}_i \times \mathbf{v}_i \rightarrow -\mathbf{m}$$

But the free energy (as a function of the order parameter) must respect these symmetries.

# GINZBURG-LANDAU THEORY I

Free energy =

interaction energy – temperature  $\times$  entropy

For  $|T - T_c| \ll T_c$  magnetization is small,  $\mathbf{M}^2 \ll 1$

Ginzburg-Landau ansatz:  $F = F_0(T) + a\mathbf{M}^2 + b\mathbf{M}^4 + \dots$

Here  $F_0(T) \equiv F(T; \mathbf{M} = 0)$

NB:  $F$  and  $\mathbf{M}$  are the free energy and magnetization *per unit volume*.

Require that: 
$$\begin{cases} T > T_c & \mathbf{M} = 0 \text{ is stable} \\ T < T_c & \mathbf{M} \neq 0 \text{ is stable} \end{cases} \Rightarrow \text{at } T = T_c : \begin{cases} a = 0 \\ b > 0 \end{cases}$$

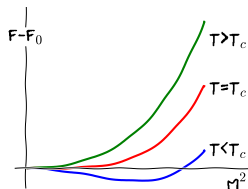
Thus, assume  $a = \alpha(T - T_c)$  and  $b = \text{const.}$

# GINZBURG-LANDAU THEORY II

$$F(T) = F_0(T) + \alpha(T - T_c) \mathbf{M}^2 + b \mathbf{M}^4$$

$$\alpha > 0, \quad b > 0$$

The *phenomenological constants*  $\alpha$  and  $b$  are input parameters in the GL theory. To fix them, use experiment or microscopic theory.



Equilibrium value of spontaneous magnetization for  $T < T_c$ :

$$0 = \frac{\partial F}{\partial M} = 2M[\alpha(T - T_c) + 2b M^2]$$

$$\Rightarrow M_{\text{eq}} = \sqrt{\alpha/2b} (T_c - T)^{1/2} \text{ —in agreement with microscopics!}$$

# GINZBURG-LANDAU THEORY III

## MAGNETIC SUSCEPTIBILITY FOR $T > T_c$ (PM PHASE)

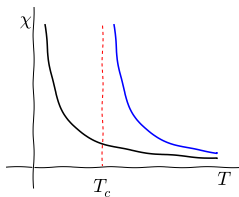
In the PM phase expect  $\mathbf{M} \uparrow \uparrow \mathbf{H}$  and  $M$  small (neglect the  $M^4$  term):

$$\begin{aligned} F &= F_0 - \mathbf{H} \cdot \mathbf{M} + \alpha(T - T_c) \mathbf{M}^2 + b \mathbf{M}^4 \\ &\approx F_0 - H M + \alpha(T - T_c) M^2 \end{aligned}$$

$$0 = \frac{\partial F}{\partial M} = -H + 2\alpha(T - T_c)M \quad \Rightarrow \quad M = \frac{1}{2\alpha} \frac{H}{T - T_c} \equiv \chi H$$

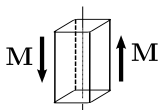
$$\chi_{\text{ferro}}(T > T_c) \propto \frac{1}{T - T_c}$$

$$\chi_{\text{Curie}} \propto \frac{1}{T}$$



# ANISOTROPY OF FERROMAGNETIC MATERIALS

Crystals have lattice symmetry (not rotationally invariant)



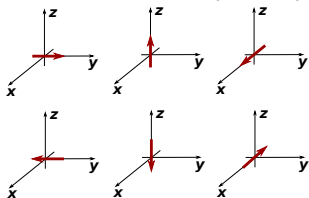
$$F - F_0 = \alpha(T - T_c) \mathbf{M}^2 + b \mathbf{M}^4 + \kappa M_z^2$$

$\kappa < 0$ : easy axis

$\kappa > 0$ : easy plane

( $\kappa \neq 0$  due to eg spin-orbit coupling)

Cubic crystals (eg, Fe): several anisotropy axes

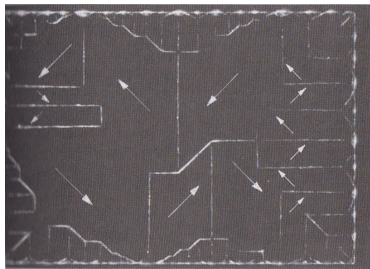


$$F - F_0 = \alpha(T - T_c) \mathbf{M}^2 + b \mathbf{M}^4 + \kappa (M_x^4 + M_y^4 + M_z^4)$$

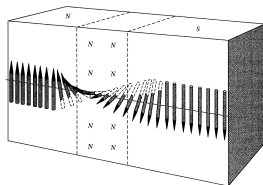
(since  $M_x^2 + M_y^2 + M_z^2 = \mathbf{M}^2$ )

# FERROMAGNETIC DOMAINS

Upon cooling a ferromagnet, spontaneous magnetization forms below  $T_c$ .



Since there are several energetically equivalent magnetization directions, specimen splits into *domains*: distant parts of the crystal are essentially independent, and phase transition develops simultaneously.



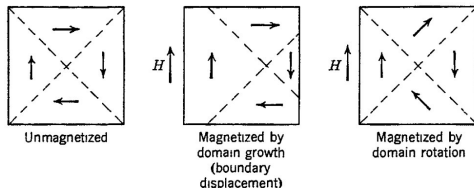
Domain walls (DWs)—the boundaries between domains—are *topological defects*: they cannot be eliminated by a smooth local changes of  $\mathbf{M}(\mathbf{r})$ .



# DOMAIN WALLS

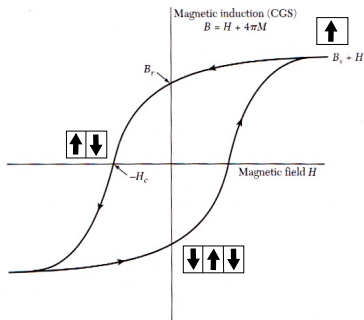
DWs are topological defects. The way of re-magnetizing the sample is to move the DWs so that unfavorably oriented domains shrink [  $\Leftrightarrow$  domain walls annihilate.]

In a field, magnetization can proceed by two mechanisms: DW displacement and/or domain rotation (usually: strong fields).



# HYSTERESIS LOOP

Crystalline defects pin the DWs.  $\Rightarrow$  moving the DWs dissipates energy. The area of the hysteresis loop (in the  $M$ - $H$ ) plane gives the total work against the friction due to defects.



**Coercivity**  $H_c$ : reduces the induction  $B$  to zero starting from **saturation**  $B_s$ . Or,  $H_{ci}$ : reduces magnetization to zero starting from **saturation**  $M_s$ . **Remanence**  $B_r$ : the value of  $B$  at  $H = 0$ .

# REPRESENTATIVE VALUES OF $T_c$ AND $M_s$

substance	Magnetization $M_s$ , gauss		$T_c$ , K	$T_{\text{melt}}$ , K
	( $T = 300$ K)	( $T = 0$ K)		
Fe	1707	1740	1043	1810
Co	1400	1446	1388	1768
Ni	485	510	627	1726
MnAs	670	870	318	1209

# HALBACH ARRAYS

J.C. Mallinson, 1973; K. Halbach, 1980: “one-sided” magnetic flux structures

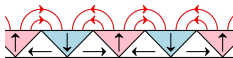


image (c) Wikipedia

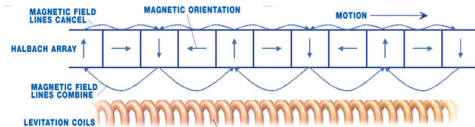


Figure 1. A Basic Halbach Array

image (c) NASA

Used in

- fridge magnets
- focusing electron beams in accelerators
- maglev levitating trains (google for “inductrack”)
- axial magnetic bearings (NASA)

# MAGNETIC FORCE MICROSCOPY (MFM)

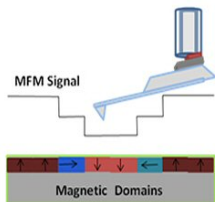
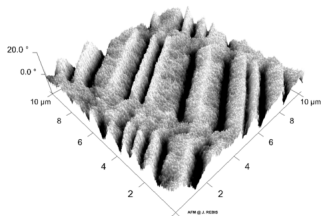


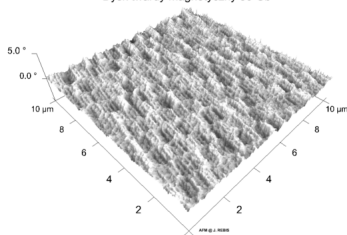
image (c) AppNano

## MAGNETIC FORCE MICROSCOPY

Dysk twardy magnetyczny 3,2 Gb



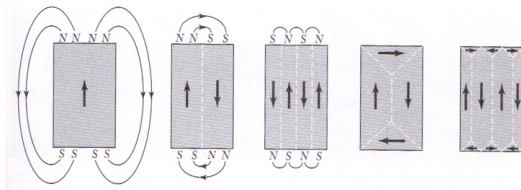
Dysk twardy magnetyczny 30 Gb



# ORIGIN OF DOMAINS

Domain structure is a consequence of an interplay between several contributions to the energy— exchange, anisotropy and magnetic — of a magnetic body.

## MAGNETIC ENERGY



magn energy

$$= \frac{1}{8\pi} \int B^2 d\Omega$$

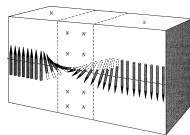
Magnetic energy of a sample with  $N$  domains is roughly  $1/N$  of an energy of a saturated sample. For two rightmost sketches: magnetic energy is zero due to the *domains of closure*.

# ORIGIN OF DOMAINS: EXCHANGE ENERGY



expensive

Bloch wall: spins tilt  
out-of-plane



much cheaper



$$\varepsilon_{12} = -J\mathbf{S}_1 \cdot \mathbf{S}_2 = -JS^2 \cos \varphi \approx -JS^2 (1 - \varphi^2/2)$$

$$\Rightarrow \text{energy cost per pair } w_{\text{ex}} = JS^2 \varphi^2/2$$

For  $N \gg 1$  spins along a chain:  $\varphi = \pi/N$

$$Nw_{\text{ex}} = NJS^2(\pi/N)^2/2 = (\pi^2/2)JS^2/N \rightarrow 0 \quad \text{as } N \rightarrow \infty$$

The wall would thicken without limit. Limiting the thickness is the *anisotropy energy*.

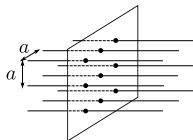
# ORIGIN OF DOMAINS: ANISOTROPY ENERGY

As a rough estimate, the anisotropy energy is proportional to the number of spins tilted away from an easy axis:

$$U_{\text{anis}} \sim K \times \text{area} \times \text{wall thickness}$$

Here  $K$  has units of energy per unit volume,  $\text{erg}/\text{cm}^3$ .

Consider a wall parallel to the face of a sc lattice and separating the domains of the opposite magnetization.



Energy per unit area:  $\sigma_{\text{wall}} = \sigma_{\text{exch}} + \sigma_{\text{anis}}$

$$\sigma_{\text{anis}} \approx KNa$$

$$\sigma_{\text{exch}} = Nw_{\text{ex}} \times 1/a^2 = (\pi^2/2)JS^2/N \times 1/a^2$$

(there are  $1/a^2$  chains per unit area)



# ORIGIN OF DOMAINS: ANISOTROPY ENERGY II

The wall energy per unit area:

$$\sigma_{\text{wall}} = \frac{\pi^2}{2} \frac{JS^2}{a^2} \frac{1}{N} + KNa$$

Minimize  $\sigma_{\text{wall}}$  with respect to  $N$ :

$$0 = \frac{\partial \sigma_{\text{wall}}}{\partial N} = -\frac{\pi^2}{2} \frac{JS^2}{a^2} \frac{1}{N^2} + Ka$$

At a minimim

$$N = \left( \frac{\pi^2}{2} \frac{JS^2}{Ka^3} \right)^{1/2} \quad \sigma_{\text{wall}} = \left( 2\pi^2 \frac{JS^2 K}{a} \right)^{1/2}$$

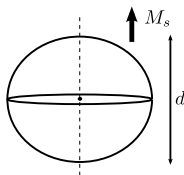
Representative values: Fe:  $N \approx 300$        $\sigma_{\text{wall}} \approx 1 \text{ erg/cm}^2$

Co:  $\sigma_{\text{wall}} \approx 3 \text{ erg/cm}^2$  (due to higher anisotropy const)

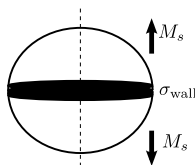
# SINGLE-DOMAIN PARTICLES

Consider a small spherical particle of a uniaxial ferromagnet.

Single domain:



Two domains, a DW in the equatorial plane:



As a very rough estimate: magnetic energy  $\sim M_s^2 \times \text{volume} \sim M_s^2 d^3$

wall energy  $\sim \sigma_{\text{wall}} \times \text{area} \sim \sigma_{\text{wall}} d^2$

Two-domain structure has twice lower magn energy.

$\Rightarrow$  the *critical* diameter is s.t.  $M_s^2 d_c^3 \sim \sigma_{\text{wall}} d_c^2$ .

For  $d < d_c = \sigma_{\text{wall}} / M_s^2$  single domain is favourable.