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## Abstract

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## 1 Introduction

There is a great deal of interests in the development of viable green technologies aimed at the enhancement of biodegradable and biocompatible polymer materials, such as the blend of poly  $\epsilon$ -caprolactone (PCL) and poly lactide (PLA) Blending is a low cost, easy implement process for modification of polymer mixture. Mixing several immiscible polymer with different physical and chemical properties has possibility to improve the overall constituents' performance.

Nowadays, the blending material of poly  $\epsilon$ -caprolactone (PCL) and poly lactide (PLA) shows a wide application in many areas. PLA has excellent biocompatibility, biodegradability and dynamic property, which shows broad prospects for tissue engineering development and other areas. But pure PLA has some drawbacks, including poor toughness, low degradation rate.

Moreover, the crystallinity of PLA is low. These drawbacks severely restrain the application of PLA material. Researchers have attempted some polymer to make blends with PLA. PHB, PCL, PP and other material which has good performance has been attempted in recent years.

Physical property improvement has been found in some researchers' blending experiments. PCL The blend of PCL and PLA shows high degradation rate, better tensile strength, which is the properties originated from PLA, while PCL with much slower degradation rate and better toughness. The blend of PCL and PLA is a promising material which can meet the requirement of environment and physical conditions. Qiaolian Lv et al.[1] used an injection molding blend of PCL and PLA. They got a maximum  $\sigma = 29.8 \pm 0.9$  MPa and  $E = 922.5 \pm 9.8$  MPa. For electrospun process, Pisani et al.[2] got a maximum  $E = 49.10 \pm 0.12$  Mpa.

The physical property of PCL-PLA blend depends on the super molecular structure, which is dominated by the crystallization condition. The melting point ( $T_m$ ) and glass transient temperature ( $T_g$ ) of PCL are far lower than those of PLA, and the crystallization temperature ( $T_c$ ) of PCL is even lower than the  $T_g$  of PLA. Someone's research manifests that the presence of minor PCL phase favors cold crystallization of PLA in their blend system because PCL is in its molten state during PLA crystallization, reducing system viscosity as a result, or acts as additional substrates.

49 The goal of this research work to improve the physical properties of the blend. Stretching of the electrospun mat  
50 induces the inner fibers to align in one direction. The unmolten PLA fibers offer crystallization loci for PCL phase.  
51 On the other hand, the molten PCL phase favors cold crystallization of PLA fibers. The present works lack this kind  
52 research method. Does this process improve the performance of blends? This question interests the author. Therefore,  
53 in this work electrospinning, stretching and hot pressing experiments are carried. Lots of morphology, microstructure,  
54 dynamic and thermal properties are characterized.

## 55 **2 Experiment**

### 56 **2.1 Sample Preparation**

57 Firstly, electrospinning process was carried out. PLA was supplied by 20 wt% PCL solution and 20 wt% PLA solution  
58 were prepared. 2 nozzles containing PCL solution and 1 nozzle containing PLA solution were used in the electrospinning  
59 process, so the mass ratio of PLA : PCL was 33:67. The electrospinning voltage was set at 8 Nev. The humidity of  
60 electrospun environment was 40%, the temperature was 25 °C. The electrospun mats were made as standard size  
61 samples.

62 On the next stage, the electrospun mat was stretched by a mechanical tester. The electrospinning mats were stretched  
63 to a series of elongation ratios. The velocity of the tensile process was 4 mm/minute, which is a low speed to avoid  
64 the fracture of the samples. 25%, 50%, 75%, 100% and 125% elongation ratio mats were made. After that the mats'  
65 double edges were fixed by heat-resistant tape in order to keep the elongation status since the mats had a rebound trend.  
66 Subsequently, the extended mats were put between two foils which stuck to 300 °C-resistant film. The mats were  
67 deposited in a heat oven which kept an 80 °C environment lasting 1 hour for PCL's melting. After 1 hour, the mats were  
68 transferred into another heat oven which kept a 30 °C environment lasting 30 minutes for isothermal crystallization.

69 Finally, the sample was tailored into a dimension of 10 mm × 5 mm, and the thickness was recorded by a film thick-  
70 ness gauge. Abundant standard sample with same length and width were prepared for the following characterizations.

## 71 2.2 Morphology and microstructure characterizations

### 72 2.2.1 Polarized Optical Microscope (POM)

73 POM is an effective facility to observe the microstructure of polymer material. With Maltese cross phenomenon in  
74 birefringence polymer crystals, the amorphous and crystallization zone can be clearly distinguished. A Linkam heating  
75 stage was used to handle the electrospinning films with the same heat treatment process as the experiment in [subsec-](#)  
76 [tion 2.1](#). The melting and the cold crystallization process of different elongation samples were observed and recorded  
77 by a Leica POM.

### 78 2.2.2 Small Angle X-ray Scattering(SAXS)

79 Small Angle X-ray Scattering(SAXS) experiment was carried on beamline BL16B1, Shanghai Synchrotron Radiation  
80 Facility. A Pilatus 2M detector( $1475 \times 1679$  pixels with a pixel size of  $172 \mu\text{m}$ ) was performed to collect the SAXS  
81 pattern. The X-ray wavelength was  $0.103 \text{ nm}$ , and the sample to detector distance was set at  $2131 \text{ mm}$ . Considering the  
82 scale of X-ray pattern on BL19U2 is much larger than the diameter of PLA fibers inside the samples. It was essential  
83 to focus the X-ray pattern. A beryllium compound reflective lens(CRL) was deployed to get a  $5 \mu\text{m}$  diameter X-ray  
84 pattern. Besides a portable POM was placed on the light path in order to determine location on blends characterized by  
85 the SAXS method.

### 86 2.2.3 Wide Angle X-ray Scattering(WAXS)

87 WAXS experiment was carried on beamline BL16B1, Shanghai Synchrotron Radiation Facility, Pilatus 2M detector  
88 ( $1475 \times 1679$  pixels with a pixel size of  $172 \mu\text{m}$ ). The X-ray wavelength was  $0.124 \text{ nm}$ , and the sample to detector  
89 distance was set at  $178.5 \text{ mm}$ .

## 90 2.3 Dynamic Property Characterization

91 The dynamic properties of the neat PCL and its blends with various annealing histories were determined by a mechanical  
92 tester at a crosshead speed of  $50 \text{ mm} \cdot \text{min}^{-1}$ ) at  $25^\circ\text{C}$  using the dog-bone shaped specimens. Strength and modulus  
93 values reported here represent an average of the results for tests run on 6 specimens.

## 94 2.4 Thermal Property Characterization

### 95 2.4.1 Thermogravimetric Analysis (TGA)

96 TGA measurements were carried out in a Mettler Toledo TGA 2 thermal analyzer. The experiments were performed  
97 under nitrogen atmosphere (flow rate of  $50 \text{ mL} \cdot \text{min}^{-1}$ ). Each sample was heated from  $0^\circ\text{C}$  to  $600^\circ\text{C}$  at  $10^\circ\text{C} \cdot \text{min}^{-1}$ .  
98 The initial degradation temperatures ( $T_0$ ) were determined at 5% mass loss, whereas temperatures at the maximum  
99 degradation rate ( $T_{max}$ ) were calculated from the first derivative of the TGA curves (DTG).

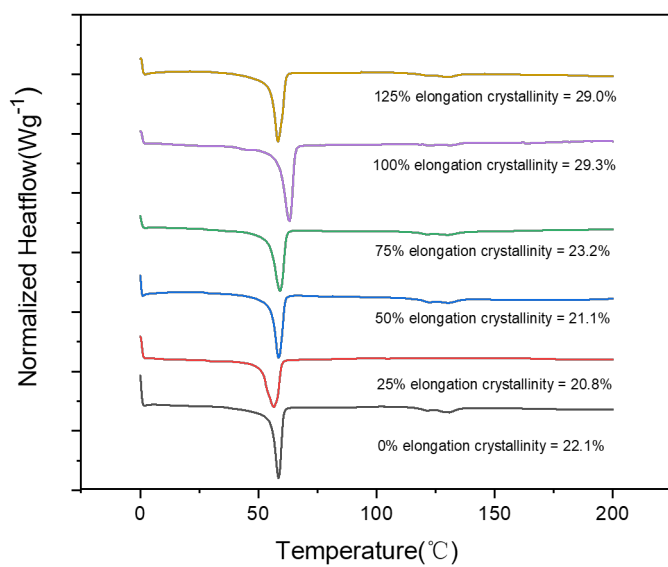
### 100 2.4.2 Differential Scanning Calorimetry (DSC)

101 DSC's experiments were performed in a Mettler Toledo DSC 3+ under nitrogen atmosphere (flow rate of  $50 \text{ mL} \cdot \text{min}^{-1}$ ).  
102 Sample weights of 2 mg were sealed in aluminum pans and heated from  $0^\circ\text{C}$  to  $200^\circ\text{C}$  at  $10^\circ\text{C} \cdot \text{min}^{-1}$ . The degree  
103 of crystallinity ( $\chi_c$ ) was calculated through [Equation 1](#)

$$\chi_c = 100\% \times \left[ \frac{\Delta H_m - \Delta_{cc}}{\Delta H_m^c} \right] \frac{1}{W_{PCL}} \quad (1)$$

## 104 3 Result and Analysis

105 As shown in, a is the melting and isothermal crystallization of 0 % elongation sample, b is the same process of 125 %  
106 elongation sample. Compare the evolution of two samples, it can be remarkably noticed that the stretching force make  
107 PLA fibers inside samples present a certain degree of orientation. During the isothermal crystallization process at  $30^\circ\text{C}$ ,  
108 the PCL grain firstly appears on the surface of PLA fiber. After a short time, the grain is fill the entire matrix.



**Table 1.** Crystallinity of components in PCL/PLA blends

Sample	Crystallinity (%) of PCL phase	Crystallinity (%) of PLA phase	Melting point(°C) of PCL phase
0%	22.1	-	58.7
25%	20.8	-	56.6
50%	21.1	-	58.6
75%	23.2	-	59.1
100%	29.3	-	63.0
125%	29.0	-	58.5

$\chi_c$  calculated using  $\Delta_m^c$  of PCL of  $139.5(\text{J} \cdot \text{g}^{-1})$

### 109 **3.1 Crystallization behavior**

## 110 **4 Conclusion**

## 111 **5 Acknowledgement**

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