

AERONAUTICS INSTITUTE OF TECHNOLOGY



Felipe Mello dos Reis

**DEVELOPMENT OF ONE-PART ALKALI-ACTIVATED
CEMENT WITH LOW-CALCIUM SOLID PRECURSORS
AND ALTERNATIVE ALKALINE SOURCES.**

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Felipe Mello dos Reis

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DEVELOPMENT OF ONE-PART ALKALI-ACTIVATED CEMENT WITH LOW-CALCIUM SOLID PRECURSORS AND ALTERNATIVE ALKALINE SOURCES.

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Resumo

Na busca por alternativas mais sustentáveis ao cimento Portland, os cimentos ativados alcalinamente têm sido amplamente estudados. Inicialmente, a maioria dos processos de mistura ocorre em duas etapas, que sacrificam a eficiência produtiva em função das melhores propriedades mecânicas. Com o objetivo aumentar a escalabilidade do processo, o desenvolvimento de sistemas monocomponentes trouxe uma tecnologia mais acessível e prática para a indústria. Ainda assim, os estudos atuais se concentram em precursores ricos em cálcio, enquanto o uso de fontes alcalinas tradicionais apresenta desafios relacionados à segurança e ao custo. Este trabalho propõe o desenvolvimento de um cimento ativado alcalinamente monocomponente utilizando precursores sólidos de baixo teor de cálcio, como sílica ativa e metacaulim, e fontes alcalinas mais seguras e acessíveis, como carbonato de potássio e hidróxido de cálcio, garantindo resistência mecânica adequada e maior viabilidade para aplicação na construção civil.

Abstract

In the search for more sustainable alternatives to Portland cement, alkali-activated cements have been extensively studied. Initially, most mixing processes occur in two steps, which sacrifice productive efficiency in favor of improved mechanical properties. Aiming to increase process scalability, the development of one-part (just-add-water) systems has brought a more accessible and practical technology to the industry. Nonetheless, current studies mainly focus on calcium-rich precursors, while the use of conventional alkaline activators raises concerns related to safety and cost. This work proposes the development of a one-part alkali-activated cement using low-calcium solid precursors, such as silica fume and metakaolin, along with safer and more affordable alkaline sources, such as potassium carbonate and calcium hydroxide. The goal is to ensure adequate mechanical performance while enhancing the viability of these materials for application in the construction industry.

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List of Abbreviations and Acronyms

CO_2	carbon dioxide
GHG	greenhouse gases
AAM	alkali-activated materials
SiO_2	silica
Al_2O_3	alumina
N-A-S-H	sodium aluminate silicate hydrate
C-A-S-H	calcium aluminate silicate hydrate
K_2CO_3	potassium carbonate
$Ca(OH)_2$	calcium hydroxide
MK	metakaolin
SF	silica fume
IPT	Institute for Technological Research
ITA	Aeronautics Institute of Technology
EDS	Energy dispersive X-ray spectroscopy
XRD	X-ray diffraction
OH^-	hydroxyl
FTIR	Fourier transform infrared spectroscopy
ABNT	Brazilian Association of Technical Standards
NBR	Brazilian Standard
ASTM	American Society for Testing and Materials
OPC	Ordinary Portland Cement
GGBFS	Ground Granulated Blast Furnace Slag
MSWIA	Municipal Solid Waste Incineration Ash
NaOH	sodium hydroxide
Na_2SiO_3	sodium silicate
KOH	potassium hydroxide
Na_2CO_3	sodium carbonate
PSD	particle size distribution
LOI	loss on ignition

MIP	mercury intrusion porosimetry
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List of Symbols

w/s Water/solids ratio
s/b Sand/binder ratio

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1 Introduction

Cement is one of the main materials in civil construction, being used from the construction of houses and buildings to bridges and highways. In developing countries such as Brazil, cement is widely used due to its low complexity and cost, which allows its use on a large scale in any location. The exponential increase in cement production, 10 times greater than the world population growth (United Nations, 1995), has been accompanied by a significant share of greenhouse gas (GHG) emissions, due to the calcination process of limestone that transforms calcium carbonate into calcium oxide and carbon dioxide in high-temperature furnaces. The production of Portland cement generates on average 842 kg of CO_2/t of clinker produced (ANDREW, 2018), representing 5% of anthropogenic GHG emissions (IEA; WBCSD, 2009).

In this context, there is a need to develop new cementitious materials that present three main properties: low GHG emissions, low cost, and high strength/durability (SCRIVENER *et al.*, 2018).

Alkaline-activated materials (AAM) - solid precursors rich in silica (SiO_2) and alumina (Al_2O_3), capable of forming binding gels composed of sodium-alumino-silicate hydrate (N-A-S-H) and calcium-alumino-silicate hydrate (C-A-S-H) - have gained prominence due to their potential to partially or totally replace Portland cement, significantly reducing the GHG emissions associated with conventional cement production.

There are two ways in which AAM can be produced: by mixing the solid precursor with a liquid alkaline activator or with a solid alkaline source and water. Two-part systems have been widely employed in the initial development of this market due to their high mechanical performance, durability, and chemical resistance. However, one-part systems are a more scalable technology due to the lower risk of handling and storing of solid activators (PROVIS, 2018).

Calcium-rich solid precursors are more used for several factors, such as the rapid strength gain (PROVIS; BERNAL, 2014), less dependence on thermal curing (KE *et al.*, 2021), and the fact that C-A-S-H reaction products tend to form a denser and less porous matrix than N-A-S-H gels (BERNAL *et al.*, 2014). In this sense, there is a technical and scientific gap in the formulation and characterization of low-calcium geopolymer monocomponents.

This work proposes the development of a one-part alkaline-activated cement focusing on low-calcium solid precursors, specifically metakaolin and silica fume, combined with safer and more accessible alternative alkaline sources, such as potassium carbonate (K_2CO_3) and calcium hydroxide ($Ca(OH)_2$). This approach aims to contribute to the formulation of more sustainable, safe, and adequately performing binders for applications in civil construction, aligning with contemporary guidelines for low environmental impact (BRASIL, 2016).

2 Literature Review

2.1 Historical Context

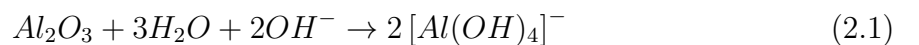
The synthesis of materials by alkali activation began in the 1930s and 1940s, when a substitute for traditional Portland cement was developed from blast furnace slag and other aluminosilicates (PACHECO-TORGAL *et al.*, 2014). From the 1970s onwards, interest in this area increased, when the French scientist Joseph Davidovits coined the term "geopolymer" and patented several formulations. His initial studies focused on the development of inorganic, non-flammable, and fire-resistant materials (PROVIS; DEVENTER, 2009).

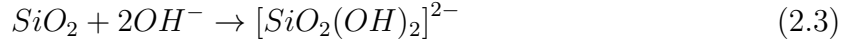
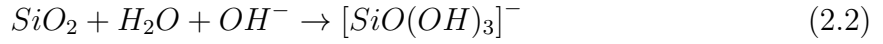
Since then, alkali-activated materials (AAM) have attracted the attention of researchers and industry due to their low energy consumption and sustainable nature (QIN *et al.*, 2022). Furthermore, as studies have advanced, AAMs have gained recognition for their mechanical properties and durability, as the polymerization reactions that occur during curing provide high compressive strength and resistance to chemical attack.

2.2 Raw Materials for AAM

2.2.1 Precursors

Precursors are materials rich in SiO_2 and Al_2O_3 that, when activated by an alkaline solution, form a three-dimensional network of aluminosilicates (RAKHIMOVA; RAKHIMOV, 2019). The mechanical and kinetic properties of AAMs are strongly influenced by the SiO_2/Al_2O_3 ratio (PROVIS, 2007). The initial activation process involves the dissolution of aluminosilicates through the breaking of covalent bonds $Si - O - Si$ and $Al - O - Al$ in a high pH environment (SEVERO *et al.*, 2013). Hydrolysis can be represented as follows:





Subsequently, the dissolved silicates and aluminates react with each other, forming a gel that undergoes polymerization and hardening processes, as illustrated in Figure 2.1.

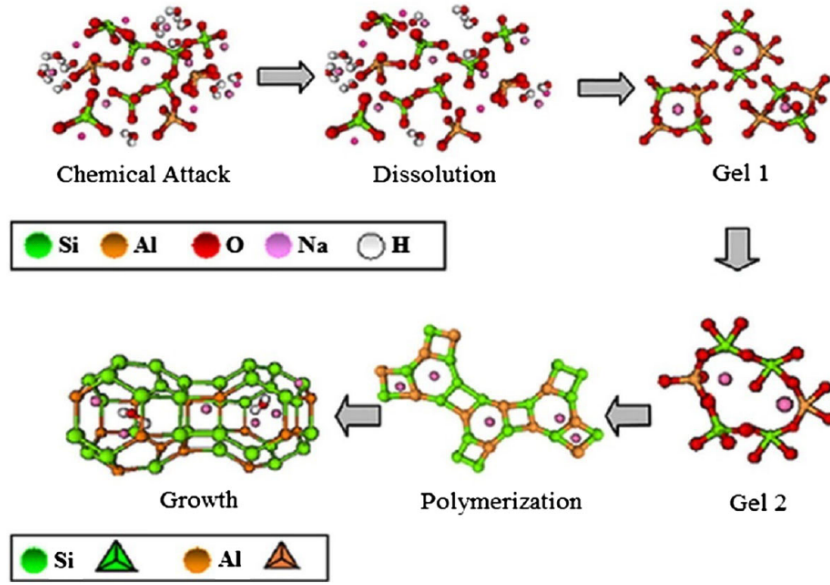


FIGURE 2.1 – Scheme of the alkaline activation process (DUXSON A. FERNÁNDEZ-JIMÉNEZ, 2006).

Precursors can be divided into two categories: those with high calcium content, such as blast furnace slag and fly ash, and those with low calcium content, such as metakaolin. Figure 2.2 shows the most common precursors and their respective chemical compositions.

The first group primarily produces calcium aluminate silicate hydrate (C-A-S-H) as a result of the activation reaction, while the second group predominantly forms sodium aluminosilicate hydrate (N-A-S-H) gel.

When the calcium levels in these precursors are high, the final product is a gel with rapid curing and high initial strength. However, these systems are more susceptible to shrinkage, cracking, and corrosion due to chloride attack. On the other hand, low-calcium systems form an tetrahedral amorphous network, which exhibits low permeability and shrinkage, better fire resistance, and a less porous structure. The $\text{SiO}_2/\text{Al}_2\text{O}_3$ ratio is responsible for the degree of polymerization of the formed gel; therefore, if the ideal ratio is not achieved, the mechanical strength and durability of the AAM may be compromised. Finally, the N-A-S-H gel requires a longer curing time and a temperature between 80 – 100 °C to reach the appropriate mechanical strength (NODEHI; TAGHVAEE, 2022).

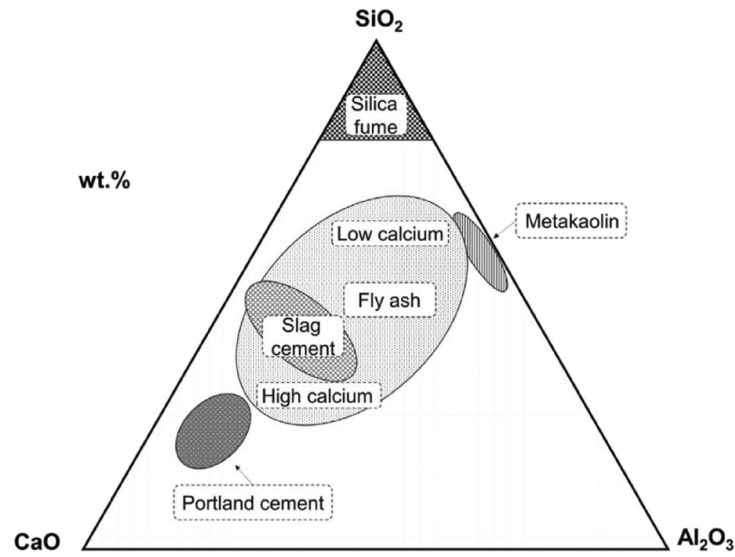


FIGURE 2.2 – Ternary diagram of the most common precursors (GIERGICZNY, 2019).

Table 2.1 presents the main characteristics of the most common precursors.

TABLE 2.1 – Characteristics of common or innovative residual materials that can be added to concrete to produce more sustainable binders (NODEHI; TAGHVAEE, 2022).

Additive name	Usual form	Average density (kg/m ³)	Average particle size (µm)	Limitations	Benefits
Portland cement (OPC)	Irregular and angular	1440	0.15–20	–	–
Silica fume	Spherical	2200	0.1–0.5	Reduces workability and initial strength	Increases compactness, mechanical strength, and durability
Ground granulated blast furnace slag (GGBFS)	Angular with rough surface	1000–1300	1.25–250	Low initial strength	Increases durability, improves ITZ, and sulfate resistance
Fly ash	Spherical	540–860	0.5–300	Low initial strength	Improves workability and long-term strength
Metakaolin	Porous, lamellar, and angular	890	1–20	Reduces workability	Fills microstructure and improves ITZ
Rice husk ash	Irregular with high porosity	504–700	5–10	Property variation and low reactivity	High silica content; improves compactness and strength
Glass powder	Irregular	2500	0.8–50	High contamination	Improves durability and pozzolanic reaction
Red mud	Irregular and needle-shaped	2700–3400	100 to over 200	High contamination	High alumina content, can improve hydration
Ceramic waste	Angular	1700	Below 100	–	Improves compactness and performance
Municipal solid waste incineration slag (MSWI)	Irregular	660–1690	–	–	Improves microstructure and reduces porosity
Paper sludge ash	Irregular	Below 100	–	–	Favourably adjusts the S/A ratio

2.2.2 Activators

The alkaline attack on the microstructure of the precursors results in the release of silicates and aluminates into the solution. The solubility of silica and alumina as a function of pH is presented in Figure 2.3.

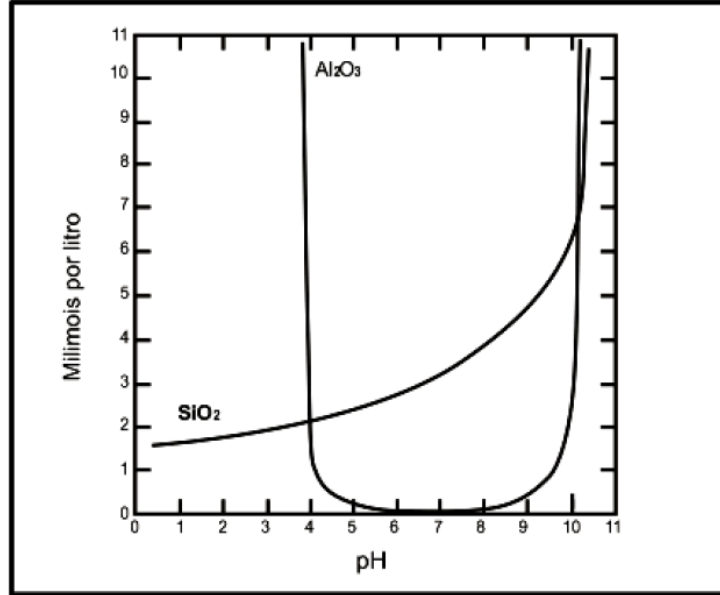


FIGURE 2.3 – Solubility of silica and alumina as a function of pH (MASON, 1952).

It is observed that the solubility of silica is low in acidic environments and high in basic media, while alumina is soluble at both extremes of pH. Therefore, for the activation reactions to occur, it is necessary that the pH of the solution is high.

Alkaline activators can be found in two forms: liquid—producing two-part geopolymer—or solid—one-part geopolymers. The main liquid alkaline activators are: sodium hydroxide ($NaOH$), sodium silicate (Na_2SiO_3), potassium hydroxide (KOH), sodium carbonate (Na_2CO_3), potassium carbonate (K_2CO_3), and potassium oxide (K_2O). The first studies on AAM focused on liquid activators, since the final product exhibits high compressive strength, adhesion, and the ability to withstand fatigue loads. In addition, they also demonstrate high resistance to freeze-thaw cycles and high temperatures (HEATH *et al.*, 2014).

Despite the advantages of two-part systems, basic solutions are corrosive and irritate human skin, making their transport and handling hazardous for workers (AWOYERA, 2019). Another point worth noting is that the production of sodium silicate occurs between 1200 – 1400 °C and emits approximately 1.514 kg of CO_2 per kg of silicate produced, in addition to significantly contributing to air pollution through dust and nitrogen and sulfur oxides (RAJAN, 2020).

Thus, one-part systems emerge as a safer alternative, since solid activators are less

hazardous and easier to handle. Even though one-part geopolymers exhibit lower mechanical strength and require thermal curing to achieve adequate performance (PROVIS, 2018), their use is more scalable.

2.3 Environmental Impacts

The main environmental advantage attributed to AAMs lies in the considerable reduction of CO_2 emissions compared to traditional Portland cement. It is estimated that the environmental impacts of solid and liquid activators are 24% and 60% of the impact caused by OPC, respectively (LUUKKENEN Z. ABDOLLAHNEJAD, 2017). Furthermore, the production of AAMs often utilizes industrial residues as raw materials, such as blast furnace slag, sewage sludge ash, rice husk ash, sugarcane straw ash, among others (MORAES *et al.*, 2024). Therefore, in addition to the valorization of industrial waste, the production of AAMs reduces the demand for natural resources from mineral deposits and provides an environmentally appropriate destination, following the guidelines of the National Solid Waste Policy (BRASIL, 2016).

2.4 Mechanical Properties

3 Methodology

3.1 Materials

For the development of one-part geopolymeric pastes and mortars, the following components were used:

- Caolin supplied by the company Brasilminas;
- Silica fume (SF) supplied by the company Elken;
- Potassium carbonate supplied by the company Neon (purity of 98%);
- Calcium hydroxide supplied by the company Neon (purity of 95%);
- Standardized quartz sand supplied by the company IPT;
- Distilled water.

The precursors used were silica fume and metakaolin. However, the purity of commercially available metakaolin is not sufficient to ensure precision in the characterization of cementitious samples. Therefore, it was produced from commercial kaolin, as detailed in Section 3.2.1.1. The alkaline sources are commercially available with high purity, so the physicochemical compositions provided by the manufacturer were used.

In addition, the quartz sand used follows the standards established by the Institute for Technological Research (IPT), as shown in Tables 3.1 and 3.2.

TABLE 3.1 – Results of physical and chemical requirements of standardized quartz sand.

Property	Result	ABNT NBR7214:2015 Requirement
Silica content (ABNT NBR14656:2001)	96.5%	$\geq 95\%$, by mass
Moisture (ABNT NBR7214:2015)	0.0%	$\leq 0.2\%$, by mass
Organic matter (ABNT NBR17053:2022)	Lighter or equal to the color of the standard solution	Color of the 2% tannic acid standard solution

TABLE 3.2 – Particle size distribution of the fractions of standardized quartz sand.

Fraction	Sieve interval	Mass percentage (%)	
		Result	ABNT NBR7214:2015 Requirement
16	(2.4 mm and 2.0 mm)	0	≤ 10
16	(2.0 mm and 1.2 mm)	97	≥ 90
30	(1.2 mm and 0.6 mm)	99	≥ 95
50	(0.6 mm and 0.3 mm)	96	≥ 95
100	(0.3 mm and 0.15 mm)	95	≥ 95

3.2 Methods

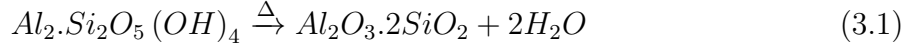
The experimental process was divided into three steps: (i) production and characterization of raw-materials; (ii) production of geopolymetric pastes and mortars; (iii) microstructural studies and mechanical tests of hardened state.

3.2.1 Production and Characterization of Raw Materials

3.2.1.1 Production of Metakaolin

Metakaolin was obtained by calcining kaolin at $700\text{ }^{\circ}\text{C}$ for 1 hour in a 200 L 18 kW laboratory furnace. The optimal calcination time and temperature were determined from preliminary tests, in which the yield of calcination and reactivity was evaluated. To ensure material homogeneity, two shallow trays with a maximum height of 10 mm were used.

The transformation of crystalline kaolinite into amorphous metakaolinite is represented by Equation 3.1.



3.2.1.2 Physicochemical Characterization of Solid Precursors

The physicochemical characterization of the solid precursors was carried out at the laboratories of the Aeronautics Institute of Technology (ITA), located in São José dos Campos-SP.

To verify the removal of hydroxyl groups (OH^-) was verified by loss on ignition (LOI). In addition, X-ray diffraction (XRD) was used to determine the crystalline phase of metakaolin and silica fume. XRD was performed by a Panalytical Empyrean diffractometer, with a 2θ interval of $10-70^\circ$, Cu-K α radiation, 0.01° step and 50 s/step.

Finally, the particle size distribution (PSD) of the solids used was determined by laser diffraction. Smaller and more irregular particles tend to have a higher specific surface area and, therefore, greater reactivity in contact with the alkaline source, which can directly influence the mechanical performance and rheological properties of the geopolymeric mortars. PSD was analyzed in a Malvern Mastersizer 3000 particle size analyzer, with air as the dispersion agent, at 1.5 bar pressure and a 40% feed rate.

3.2.2 Production of Geopolymeric Pastes and Mortars

3.2.2.1 Mix Design

The development of one-part geopolymeric mortars followed a systematic experimental design, aiming to evaluate the effect of different compositions on physicochemical and mechanical properties.

The variables considered in the study were:

- Proportion between solid precursors (metakaolin and silica fume);
- Content of alkaline activators (K_2CO_3 and $Ca(OH)_2$);
- Water/solids ratio (w/s);
- Sand/binder ratio (s/b).

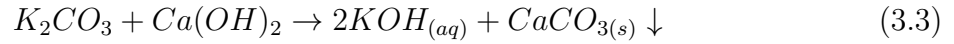
The variable of interest in this experiment is the Si/Al ratio, which will be varied from 1.0 to 5.0, calculated based on the proportion of metakaolin and silica fume.

Initially, the water/solids ratio was determined by ensuring adequate workability of the pastes.

In addition, due to stoichiometric balance, the Al/K ratio will be constant and equal to 1, according to the empirical formula on Equation 3.2 (DAVIDOVITS, 1991), where M is a sodium or potassium cation.

$$M_n \{(SiO_2)_z AlO_2\}_n \cdot wH_2O \quad (3.2)$$

Furthermore, the K/Ca ratio will be constant and equal to 2, respecting the precipitation reaction of potassium carbonate with calcium hydroxide, as shown in Equation 3.3.



With the paste proportions well defined, the production of mortars maintained a 1:2 ratio between binder and sand, analogously in the previous researches (BATISTA *et al.*, 2025).

Table 3.3 presents the different formulations produced, with the respective mass proportions of the components.

TABLE 3.3 – Compositions of the produced geopolymeric mortars.

Sample	Precursors (%)		Activators (%)		w/s	s/b	Si/Al
	MK	SF	K_2CO_3	$Ca(OH)_2$			
A1	XX	XX	X,X	X,X	X,XX	2	1,0
A2	XX	XX	X,X	X,X	X,XX	2	2,0
A3	XX	XX	X,X	X,X	X,XX	2	3,0
A4	XX	XX	X,X	X,X	X,XX	2	4,0
A5	XX	XX	X,X	X,X	X,XX	2	5,0

3.2.2.2 Mixing Procedure

The production of the mixtures followed standardized procedures. For the production of mortars and compressive strength testing, the procedures of the Brazilian standard (ABNT, 2019) were followed. For the production of pastes, the American standard (ASTM, 2006) was chosen, since the Brazilian standard does not specify the mixing procedure for cementitious pastes without fine aggregate. Both procedures were adapted for the preparation of small-volume samples.

3.2.2.3 Molding and Curing of Specimens

For the compressive strength test, the specimens were prepared in prismatic molds with dimensions $40 \times 40 \times 40 \text{ mm}$, previously lubricated with oil-based release agent.

For each composition, 9 specimens were molded, intended for testing at ages of 1, 3, and 7 days (3 specimens for each age). It was not necessary to perform tests at 28 days, as the thermal curing of the binders used provides high initial strength gain.

Thermal curing was carried out in an oven maintained at $(60 \pm 2)^\circ\text{C}$ and a minimum relative humidity of 95% for 24 hours, as recommended by the standard (ABNT, 2006), to ensure the activation of the binders and accelerate the curing process. It is noteworthy that demolding was performed 24 hours after the start of the curing process.

Demolding was performed 24 hours after molding, and the specimens were immediately transferred to the corresponding curing conditions until the testing age.

For microstructural analyses, small samples were separated, with hydration interrupted by immersion in ethyl alcohol and vacuum filtration, followed by drying in an oven at 40°C for 24 hours. These samples were stored in hermetically sealed containers to prevent rehydration.

3.2.3 Microstructural studies and Mechanical tests of hardened state

3.3 Characterization of Geopolymeric pastes

Energy-dispersive X-ray spectroscopy (EDS) was used to determine the proportion of chemical elements, performed together with scanning electron microscopy (SEM) to evaluate the morphology of the solid precursors. SEM/EDS was performed using a TESCAN VEGA 3 XMU device and Oxford EDS 133 eV detector, with gold coated.

To verify the presence of $Al - Si - O$ bonds, Fourier-transform infrared spectroscopy (FTIR) was carried out in a PerkinElmer spectrophotometer, with a spectrum range of $4000 - 400 \text{ cm}^{-1}$ and a spectral resolution of 1 cm^{-1} . Moreover, XRD and PSD were performed under the same conditions applied for the raw materials.

Lastly, the total porosity and pore size distribution of the hardened pastes were determined by mercury intrusion porosimetry (MIP).

3.3.0.1 Compressive Strength of mortars

For the compressive strength test, the specimens were placed in a hydraulic press with a 600 kN-load limit, applying load at a rate of 0.5 kN/s until failure. The strength was

calculated by the equation:

$$R_c = \frac{F_c}{A_t} \quad (3.4)$$

Where:

- R_c is the compressive strength, in MPa;
- F_c is the maximum applied load, in N;
- A_t is the cross-sectional area, in mm².

For statistical analysis, the Tukey test was performed, allowing the identification of significant differences between sample groups, considering a significance level of 95%.

4 Results and Discussion

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5 Conclusion

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6. AUTOR(ES): Felipe Mello dos Reis			
7. INSTITUIÇÃO(ÕES)/ÓRGÃO(S) INTERNO(S)/DIVISÃO(ÕES): Aeronautics Institute of Technology – ITA			
8. PALAVRAS-CHAVE SUGERIDAS PELO AUTOR: AAM; Geopolymer; One-part			
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11. RESUMO: Na busca por alternativas mais sustentáveis ao cimento Portland, os cimentos ativados alcalinamente têm sido amplamente estudados. Inicialmente, a maioria dos processos de mistura ocorre em duas etapas, que sacrificam a eficiência produtiva em função das melhores propriedades mecânicas. Com o objetivo aumentar a escalabilidade do processo, o desenvolvimento de sistemas monocomponentes trouxe uma tecnologia mais acessível e prática para a indústria. Ainda assim, os estudos atuais se concentram em precursores ricos em cálcio, enquanto o uso de fontes alcalinas tradicionais apresenta desafios relacionados à segurança e ao custo. Este trabalho propõe o desenvolvimento de um cimento ativado alcalinamente monocomponente utilizando precursores sólidos de baixo teor de cálcio, como sílica ativa e metacaulim, e fontes alcalinas mais seguras e acessíveis, como carbonato de potássio e hidróxido de cálcio, garantindo resistência mecânica adequada e maior viabilidade para aplicação na construção civil.			
12. GRAU DE SIGILO: (X) OSTENSIVO () RESERVADO () SECRETO			