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Ultrathin MoS₂-coated carbon nanospheres as highly efficient electrocatalysts for hydrogen evolution reaction

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ABSTRACT

The ultrathin MoS₂-coated carbon nanospheres have been prepared by a facile solvothermal method using acid-treated carbon nanospheres (ATCNS) as support. TEM images show that ATCNS are homogeneously coated by MoS₂ layers. The thickness of MoS₂ coating on ATCNS is about 5 nm. XRD data confirm that the few stacking layers and low crystalline of MoS₂ nanosheets on ATCNS, which could provide more active sites for hydrogen evolution reaction (HER). The electrocatalytic activity and stability of MoS₂/ATCNS for HER have been investigated. The results show that MoS₂/ATCNS has better electrocatalytic activity for HER than pure MoS₂. It can be speculated that ATCNS play crucial role in increasing conductivity and electrocatalytic activity for HER of MoS₂/ATCNS. The nanostructure of ultrathin MoS₂ coating on ATCNS is a promising electrocatalysts for HER.

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Introduction

Hydrogen has been pursued to be one of the most promising energy carriers for the effective conversion and utilization of the sustainable resources such as solar energy, wind energy and waterpower [1–3]. Recently, water splitting using electrochemistry [4–7] or photoelectrochemistry cells [8–12] has been widely investigated as valid ways for hydrogen evolution reaction (HER). The most efficient electrocatalysts for water splitting are noble metals, such as Pt.

The obvious disadvantage of noble metals is the high price and limited reserves which are big barriers for widely industrial utilization [13]. Therefore, the exploration of high efficient HER electrocatalysts from earth-abundant elements has become a global hot research. In fact, various noble-metal-free catalysts including transition metal sulfides [14,15], oxides [16], and compounds [17,18] have been widely developed to partially substitute the role of Pt in water splitting. Some substitutes with rich earth-abundance have demonstrated high activity close to that of Pt, but there are still some challenges to improve their catalytic

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efficiency and stability to meet the practical requirements of industrial application [19].

Among these candidates, molybdenum disulfide (MoS_2) nanomaterials have showed good electrocatalytic activity for HER, which has been confirmed by the computational and experimental results [20,21]. The active sites of MoS_2 are mainly the rims and edges of MoS_2 nanosheets rather than the basal planes [22], and it is proportional to the electrocatalytic activity for HER. However, as a typical two-dimensional (2D) transition-metal sulfide, MoS_2 has analogous layered structure of graphene, which are vulnerable to result in severe stacking and aggregation owing to the high surface energy and interlayer van der Waals attraction [23]. The obvious stacking and aggregation phenomena will decrease the number of the active sites for HER. Another disadvantage of MoS_2 is the poor conductivity because of the semi-conductive nature, in which the electrons prefer to transfer along the lamella structure of MoS_2 nanosheets. Therefore, designing suitable nanostructures of MoS_2 with the more exposed active sites, as well as improving of the conductivity of MoS_2 by using ideal supports, is the key to improve the properties of MoS_2 -based electrocatalysts for HER effectively.

Owing to the excellent conductivity and stability, carbon nanomaterials as support for MoS_2 -based electrocatalysts have been widely studied including graphene [24,25], reduced graphene oxide [26,27], carbon nanotubes [28], carbon nanospheres (CNS) [29]. These researches have confirmed that carbon nanomaterials could improve the exposed active sites and conductivity of MoS_2 . However, there remain some problems such as cost barrier, complicated procedures, aggregation growth of MoS_2 etc. In this work, a facile solvothermal method using situ hydrolysis of ammonium thiomolybdate ($(\text{NH}_4)_2\text{MoS}_4$) in N, N-dimethylformamide (DMF) has been developed to prepare ultrathin MoS_2 -coated acid-treated carbon nanospheres ($\text{MoS}_2/\text{ATCNS}$) with highly homogenous dispersion and low crystalline structure. The results show that as-prepared ultrathin $\text{MoS}_2/\text{ATCNS}$ nanostructures have the enhanced electrocatalytic activity and stability for HER. The effect of ATCNS and situ hydrolysis of $(\text{NH}_4)_2\text{MoS}_4$ on the morphology and structure of $\text{MoS}_2/\text{ATCNS}$ have been discussed.

Experimental

Synthesis of catalysts

Firstly, 80 mg of the acid-treated carbon nanospheres and 80 mg $(\text{NH}_4)_2\text{MoS}_4$ were dissolved in the solution of 64 mL DMF and 16 mL deionized water under stirring for 1 h and sonication for 1 h. Then the mixture and 0.80 mL hydrazine was transferred into a Teflon stainless steel autoclave. The solvothermal reaction was carried out at the temperature of 200 °C for 12 h. The as-prepared $\text{MoS}_2/\text{ATCNS}$ were collected by centrifugation and subsequently washed with deionized water and absolute ethanol three times. The products were obtained and dried at 60 °C for 24 h in a vacuum oven. For comparison, the pure MoS_2 and MoS_2 /original carbon nanospheres (MoS_2/OCNS) samples have also been synthesized under identical conditions.

Catalysts characterization

Crystallographic structure of all as-prepared samples was investigated with X-ray powder diffraction (XRD, X'Pert PRO MPD, Cu KR) at a scanning rate of 1 °C min⁻¹. XRD data were collected in the 2 θ ranges from 5 to 80°. The morphology of the samples was examined with field-emission scanning electron microscopy (SEM, Hitachi, S-4800). Transmission electron microscopy (TEM) images were collected on HRTEM, JEM-2100UHR with an accelerating voltage of 200 kV. The samples were prepared by dropping the ethanol solution of samples on the Cu grids and were observed at 100 kV.

Electrocatalytic measurements for HER

The glassy carbon electrode (GCE) as work electrode with the diameter of 4.0 mm (Gamry Reference 600 Instruments, USA) was polished with alumina slurry and cleaned with ethanol and DI water. A conventional three-electrode system was used with Ag/AgCl as the reference electrode and Pt foil as counter electrode. The potential values are corrected to the reverse hydrogen electrode (RHE) according the equation $E(\text{RHE}) = E(\text{Ag/AgCl}) + 0.22 \text{ V}$. Typically, 4 mg of sample and 20 μL Nafion solution (5 wt%) were dispersed in 1 mL water–ethanol solution with volume ratio of 1:1 by sonicating for 1 h to form a homogeneous ink. Then 10 μL of the dispersion was loaded onto a glassy carbon electrode. The cathodic current density was calculated by the geometric area of GCE which is 0.1256 cm². Prior to each electrochemical measurement, the electrolyte solution was purified with N_2 for 30 min to remove the dissolved oxygen, and stable polarization curves were recorded after 10 cycles.

Results and discussion

Fig. 1 shows FT-IR spectra of the original and acid-treated carbon nanospheres. Fig. 1a shows that there are only a few oxygen-containing functional groups on the surface of OCNS. In Fig. 1b, the peak of 3400 cm⁻¹ belongs to hydroxy group. The peaks of 1718, 1583 and 1397 cm⁻¹ correspond to carboxyl group on the surface of ATCNS. The large amount of oxygen-containing functional groups on the surface of ATCNS may be very useful for the homogeneous interaction and decreasing stacking layers of $\text{MoS}_2/\text{ATCNS}$ [24].

Fig. 2 shows SEM images of carbon nanospheres after acid-treatment. The homogeneous morphology and uniform size of ATCNS have also been confirmed. The pure ATCNS with the diameter of about 40 nm has good dispersion, which are very suitable for fabricating $\text{MoS}_2/\text{ATCNS}$. SEM images of the original carbon nanospheres (in Fig. S1) reveal that there is no obvious difference in morphology compared with ATCNS, which implies that the acid-treatment does not change the morphology and dispersion of the carbon nanospheres.

Fig. 3 shows the TEM images of ATCNS and $\text{MoS}_2/\text{ATCNS}$. Fig. 3a and b shows that the pure ATCNS are highly uniform and well dispersed with the diameter of about 40 nm, which is consistent with the results observed in Fig. 2. Fig. 3c and d shows that $\text{MoS}_2/\text{ATCNS}$ have uniformly spherical morphology. Fig. 3c indicates that ATCNS is uniformly coated

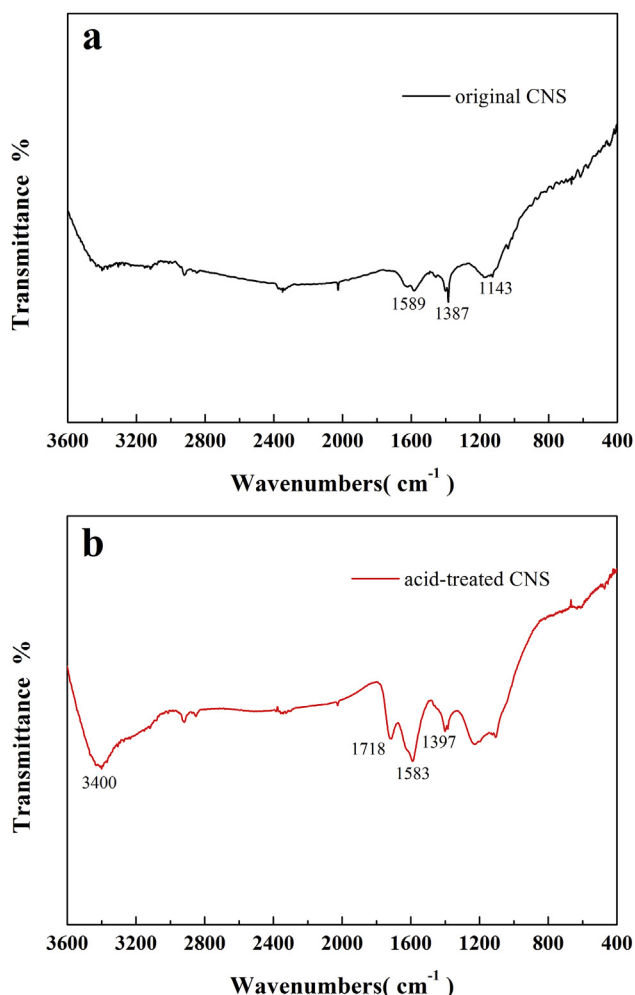


Fig. 1 – FT-IR spectra of the carbon nanospheres. (a) original carbon nanospheres, (b) acid-treated carbon nanospheres.

with MoS₂ nanosheets. And the diameter of MoS₂/ATCNS increases to about 50 nm. So the thickness of MoS₂ coating was estimated to be about 5 nm according to the 40 nm diameter of ATCNS, which implies that the staking of MoS₂ on ATCNS has not been severe. Under higher magnification, Fig. 3d shows the ultrathin MoS₂ coating on ATCNS is homogeneous and tight coupling on ATCNS. The aggregation structures of MoS₂ are not observed, which confirms the homogeneous conjunction and few stacking layers of MoS₂. Fig. 3e confirms the thickness and close interaction of ultrathin MoS₂ coating on ATCNS. As shown in Fig. 3f, the ultrathin MoS₂ coating on ATCNS are composed of a few stacking layers (5–8 layers) with the increasing 002 interlayer distance of 0.745 nm, which means low crystalline and weak growth along 002 plane. The homogeneous interaction and decreasing stacking layers of MoS₂/ATCNS may be attributed to the large amount of hydroxy and carboxyl group on the surface of ATCNS. In addition, the situ hydrolysis of (NH₄)₂MoS₄ can release H₂S as the sulfur source for the fewer stacking and controllable growth of MoS₂ nanosheets [30,31]. Meanwhile, the effect of DMF and water as solvent on the morphology of MoS₂/ATCNS can be helpful for the good dispersion and controllable growth of

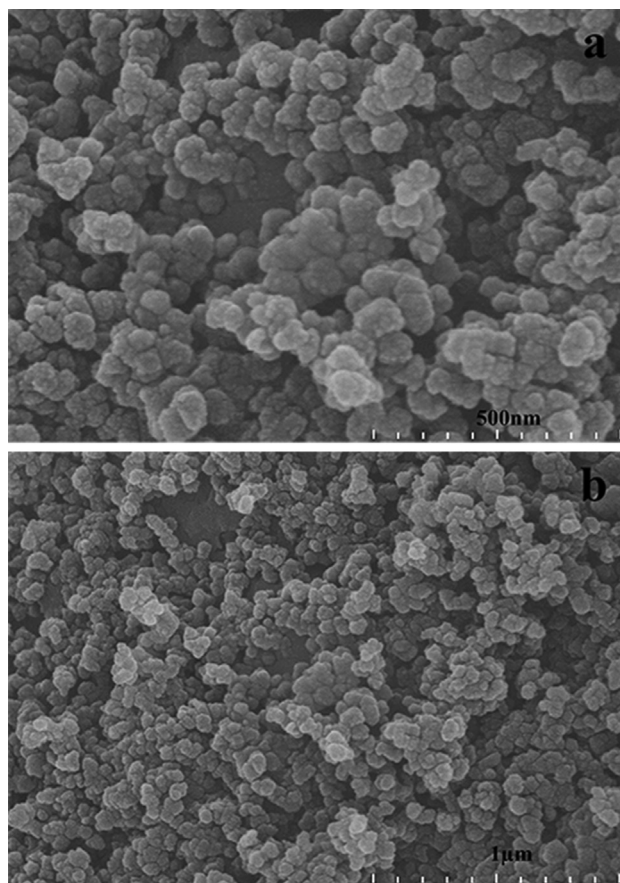


Fig. 2 – SEM images of carbon nanospheres after acid-treatment.

MoS₂ nanosheets [28]. For comparison, the as-prepared pure MoS₂ under identical conditions has severe aggregation and many stacking layers (in Fig. S2), which indicating the less active sites and weak activity for HER. From SEM images of MoS₂/ATCNS (in Fig. S3), the obvious aggregation of MoS₂ can't be observed. The stacking layers of MoS₂ are very small, which imply that MoS₂ have better dispersion on ATCNS. Ultrathin layers of MoS₂ with more catalytic active sites have a very good coating on the surface of ATCNS. Therefore, it is meaningful to use ATCNS as support for improving the good dispersion and the increasing the numbers of catalytic active sites of MoS₂.

Fig. 4 shows XRD patterns of pure ATCNS, pure MoS₂ and MoS₂/ATCNS, respectively. The peaks at 25.4° and 43.5° are indexed to (002) and (100) planes of ATCNS in MoS₂/ATCNS, respectively. The peaks at 31.8° and 56.7° are indexed to (100) and (110) planes of MoS₂ in MoS₂/ATCNS, respectively (JCPDS card no.37-1492). No other peaks from impurities have been detected, indicating the high purity of MoS₂/ATCNS. In comparison to the pure MoS₂, the (002) peak of 14.5° of MoS₂ in MoS₂/ATCNS is almost missing, which means the lower crystalline and fewer stacking of MoS₂ nanosheets. In addition, the weak peak of 9.2° for MoS₂/ATCNS means that ATCNS may insert into the MoS₂ layers and expand the interlayer distance of MoS₂ [32], which can be confirmed by Fig. 3f. The less stacking layers and weaker intensity of 002 could provide more rims and edges of MoS₂ as active sites for HER [33].

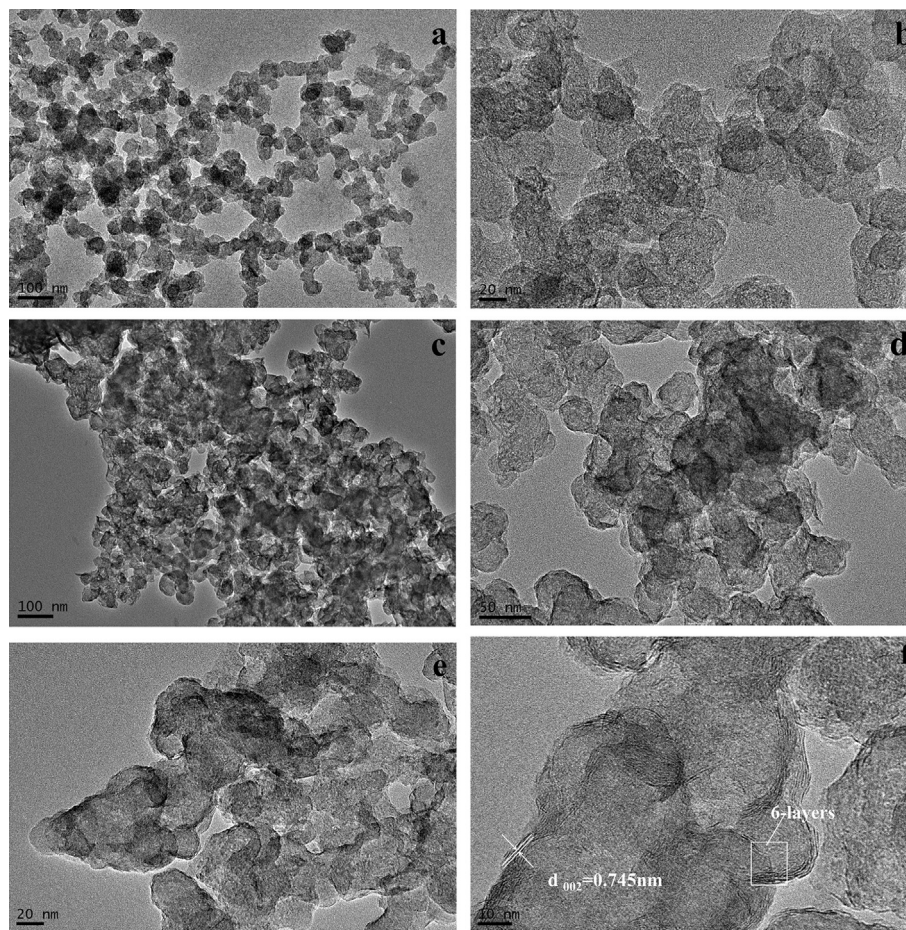


Fig. 3 – (a), (b) TEM images of acid-treated carbon nanospheres. (c), (d), (e) and (f) TEM images of MoS₂/ATCNS.

To demonstrate the electrocatalytic properties of MoS₂/ATCNS for HER, the linear sweep voltammograms (LSV) and tafel plots of the different samples were performed in 0.5 M H₂SO₄ solution (as shown in Fig. 5A). Fig. 5A shows the polarization curves for the different samples including bare

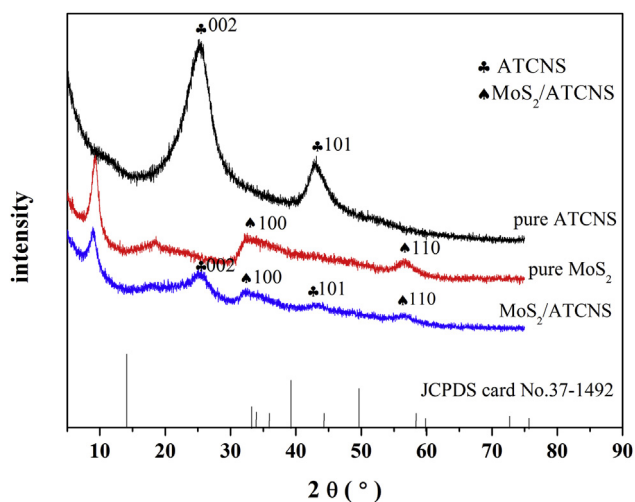


Fig. 4 – XRD patterns of different samples: pure ATCNS, pure MoS₂ and MoS₂/ATCNS.

glass carbon electrode (GCE), pure MoS₂, MoS₂/ATCNS, MoS₂/OCNS and physical mixture of MoS₂ and ATCNS (MoS₂-ATCNS). It can be observed that MoS₂/ATCNS exhibits good electrocatalytic activity for HER with onset potential of -0.14 V (vs. RHE) and high current densities of ~ 10 and ~ 50 mA cm⁻² at overpotentials of 200 and 240 mV. This onset potential of MoS₂/ATCNS is smaller than that of pure MoS₂, MoS₂/OCNS and MoS₂-ATCNS, which suggests the better electrocatalytic activity for HER. The observed cathodic current densities for pure MoS₂, MoS₂/OCNS MoS₂-ATCNS were much lower. Because the cathodic current density is proportional to the quantity of evolved hydrogen [34], the larger current density of MoS₂/ATCNS indicates the better HER electrocatalytic behaviour, which may be attributed to the improved conductivity, the homogeneous interaction and ultrathin MoS₂ coating on the surface of ATCNS.

The ideal electrocatalysts should have low Tafel slopes and high cathodic current densities. Pt has a high current density in the order of 10^{-3} A cm⁻² and a Tafel slope of 30 mV dec⁻¹ for HER [5]. Fig. 5B shows Tafel slopes of 53, 54 and 55 mV dec⁻¹ for MoS₂/ATCNS, MoS₂-ATCNS and MoS₂/OCNS, respectively. MoS₂/ATCNS display the best electrocatalytic activity with the smallest slope value of 53 mV dec⁻¹, which is similar to the previous report [35]. For HER mechanism, the first step is a fast discharge reaction (Equation (1)) and the rate determining step is the electrochemical desorption of H_{ads} and H₃O⁺ to form

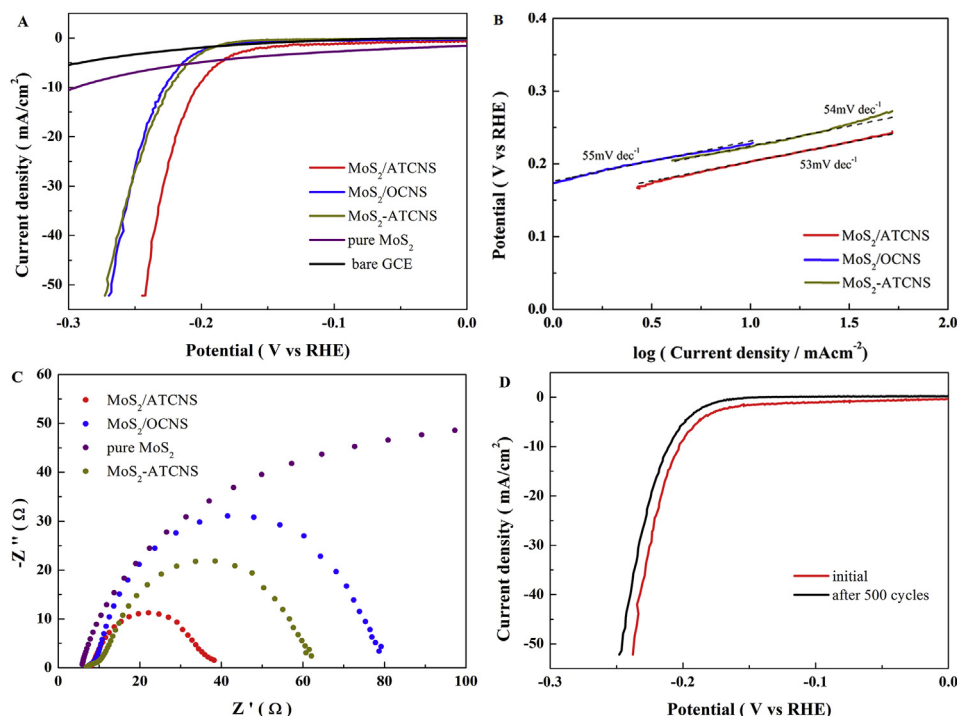
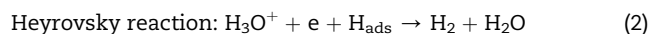
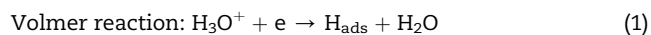


Fig. 5 – (A) Linear sweep voltammogram curves in 0.5 M H₂SO₄ for different electrocatalysts. The scan rate is 20 mV s⁻¹ and the scan region ranges from 0 to -0.3 V vs RHE, (B) Tafel plots of different electrocatalysts, (C) AC impedance spectra for different electrocatalysts at overpotential of 0.2 V from 10⁵ to 0.1 Hz with an AC voltage of 5 mV, (D) Stability for the MoS₂/ATCNS composite modified electrode with initial polarization curve (red curve) and after 500 cycles (black curve). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

hydrogen (Equation (2)), which complies with Volmer–Heyrovsky mechanism [36,37].



The conductivity of different samples can be measured by AC impedance (in Fig. 5C). MoS₂/ATCNS shows much smaller charge-transfer resistance than pure MoS₂, MoS₂/OCNS and MoS₂-ATCNS. The enhancement of conductivity also benefits from the homogeneous conjunction between ATCNS and MoS₂. The smaller resistance means much faster electron transfer and improved efficiency in HER.

The long-term stability of electrocatalysts is an important requirement for industrial application. A long-term potential sweep of MoS₂/ATCNS was performed from -0.3 to 0 V vs. RHE (Fig. 5D). The LSV retained similar current density to the initial one and a slight loss could be observed after 500 cycles, indicating that MoS₂/ATCNS has good stability for HER.

From FR-IT data, ATCNS have more oxygen-containing functional groups, which is very useful for improving the growing of ultrathin MoS₂ on carbon nanospheres (in Fig. 3). The good coating of MoS₂/ATCNS means better conductivity. A few stacking layers of MoS₂/ATCNS would provide more active sites for HER. Therefore, MoS₂/ATCNS sample should

have the better HER activity. To obtain further insight into the electrocatalytic activity of MoS₂/ATCNS, exchange current densities of various catalysts have been investigated (Fig. 6). The exchange current densities of various samples were calculated by using extrapolation methods (Table 1). The resulting exchange current densities of MoS₂/ATCNS samples is 2.57 μA cm⁻², which is larger than 0.65 μA cm⁻² of MoS₂/OCNS and 0.68 μA cm⁻² of MoS₂-ATCNS. The exchange

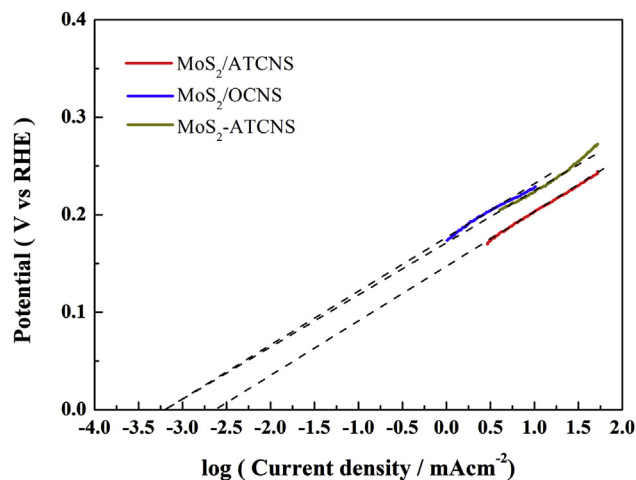


Fig. 6 – Calculated exchange current densities of various samples by using extrapolation methods.

Table 1 – Calculations of the exchange current densities of various samples.

Materials	Log (Current density/ mA cm ⁻²) at $\eta = 0$ V	Exchange current densities j_0 (μ A/cm ²)
MoS ₂ / ATCNS	–2.59	2.57
MoS ₂ / OCNS	–3.19	0.65
MoS ₂ - ATCNS	–3.17	0.68

current densities confirmed that MoS₂/ATCNS have the better electrocatalytic behaviour for HER.

Conclusion

In conclusions, ultrathin MoS₂/ATCNS with homogenous dispersion and close interaction has been fabricated by a facile solvothermal method. The TEM showed the thickness of ultrathin MoS₂ coating on ATCNS is about 5 nm. XRD confirmed the few stacking layers and low crystalline of MoS₂ nanosheets, which could provide more active sites for HER. It can be speculated that both ATCNS and situ hydrolysis of (NH₄)₂MoS₄ play crucial role in improving conductivity and preparing ultrathin MoS₂ coating of MoS₂/ATCNS. The good conductivity means the fast electron transport between the active site and the electrode. The few stacking layers mean that there are more exposed active sites on the surface of MoS₂/ATCNS. According to the LSV and exchange current densities results, MoS₂/ATCNS have the better electrocatalytic activity for HER. Therefore, the nanostructure of ultrathin MoS₂ coating on ATCNS is a promising choice to prepare highly efficient electrocatalysts for HER.

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Appendix A. Supplementary data

Supplementary data related to this article can be found at <http://dx.doi.org/10.1016/j.ijhydene.2015.03.150>.

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