



FRITZ-HABER-INSTITUT  
MAX-PLANCK-GESELLSCHAFT



# FHI-vibes

## A Toolkit for Finding and Analyzing Thermal Insulators

ASE Workshop

Chalmers University, November 19

**Florian Knoop**, Tom Purcell, Maja-Olivia Lenz,  
Matthias Scheffler, and Christian Carbogno



# Warmup



Florian Knoop

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PhD student at FHI

[gitlab.com/flokno](https://gitlab.com/flokno)

PhD at FHI in Berlin, started January 2017:

*ab initio* Thermal Transport

Tools for High-Throughput Screening

# Warmup



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PhD student at FHI

Current Project:

[gitlab.com/flokno](https://gitlab.com/flokno)

Compute thermal conductivity for 100 materials

→ Starting point for developing **data science perspective** on  
lattice thermal conductivity (non-metals)

# Outline

## Part 1 — Physics

**Thermal conductivity** by *ab initio* Green-Kubo

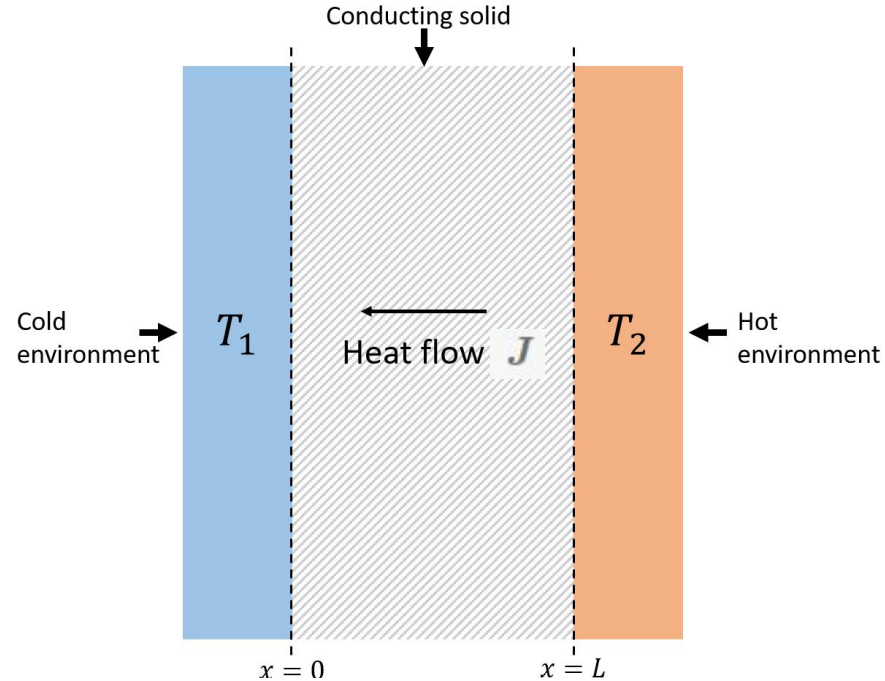
High-Throughput estimation of **Anharmonicity**

## Part 2 — Tooling

**FHI-vibes**

Outlook

# Thermal Transport



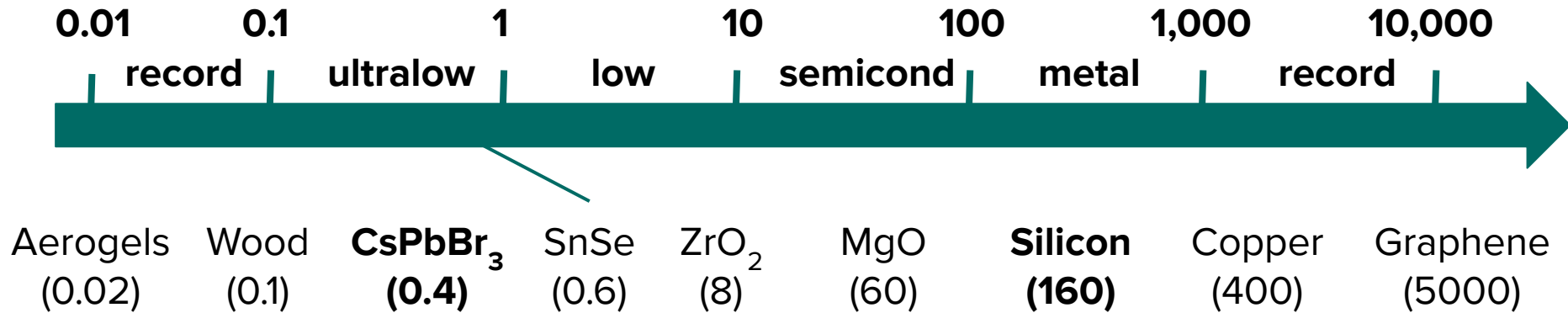
Fourier's law:

$$\mathbf{J} = -\kappa \nabla T$$

# Thermal Conductivity

Unit:

W / m K [**Watt** per **meter Kelvin**]



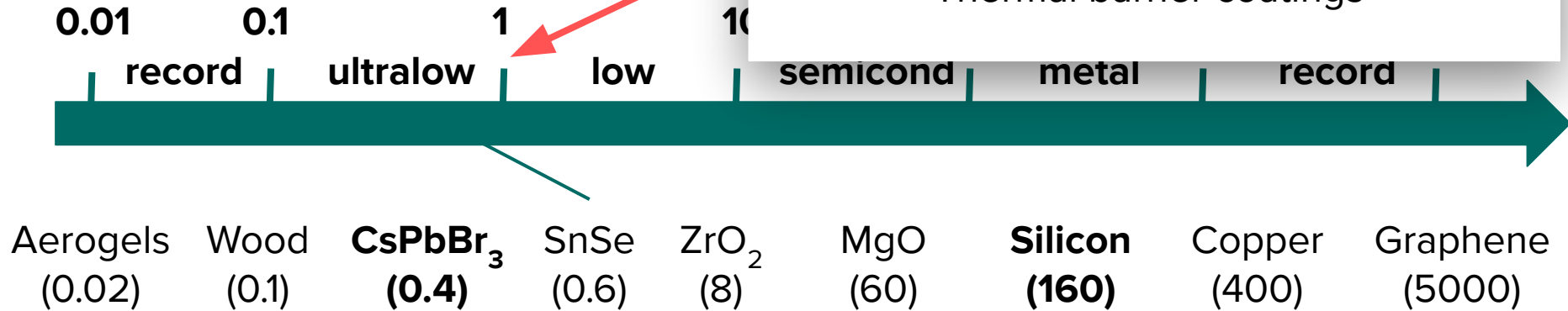
# Thermal Conductivity

Unit:

W / m K [**Watt** per **meter Kelvin**]

Applications for **Thermal Insulators**:

- Thermoelectrics
- Thermal barrier coatings



# *Ab initio* Green-Kubo



Real space  
real time  
**heat flux**

$$\mathbf{J}(t) = \sum_I \sigma_I \dot{\mathbf{R}}_I(t)$$

**Atomic velocities**  
from *ab initio*  
Molecular Dynamics

$$\kappa \propto \int d\tau \langle \mathbf{J} \mathbf{J} \rangle (\tau)$$

**Atomic stress**  
contribution  
`atoms.get_stresses()`

**No approximations to potential energy surface**

C. Carbogno, R. Ramprasad, M. Scheffler, *PRL* **118**, 175901 (2017)



# *Ab initio* Green-Kubo



**Fluctuation Dissipation  
Theorem**

$$\mathbf{J}(t) = \sum_I \sigma_I \dot{\mathbf{R}}_I(t)$$



$$\kappa \propto \int d\tau \langle \mathbf{J} \mathbf{J} \rangle (\tau)$$

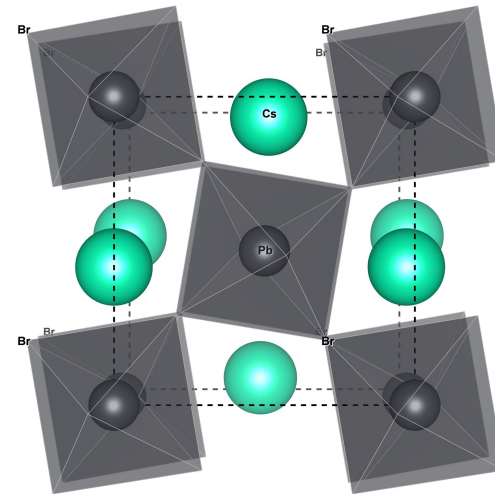
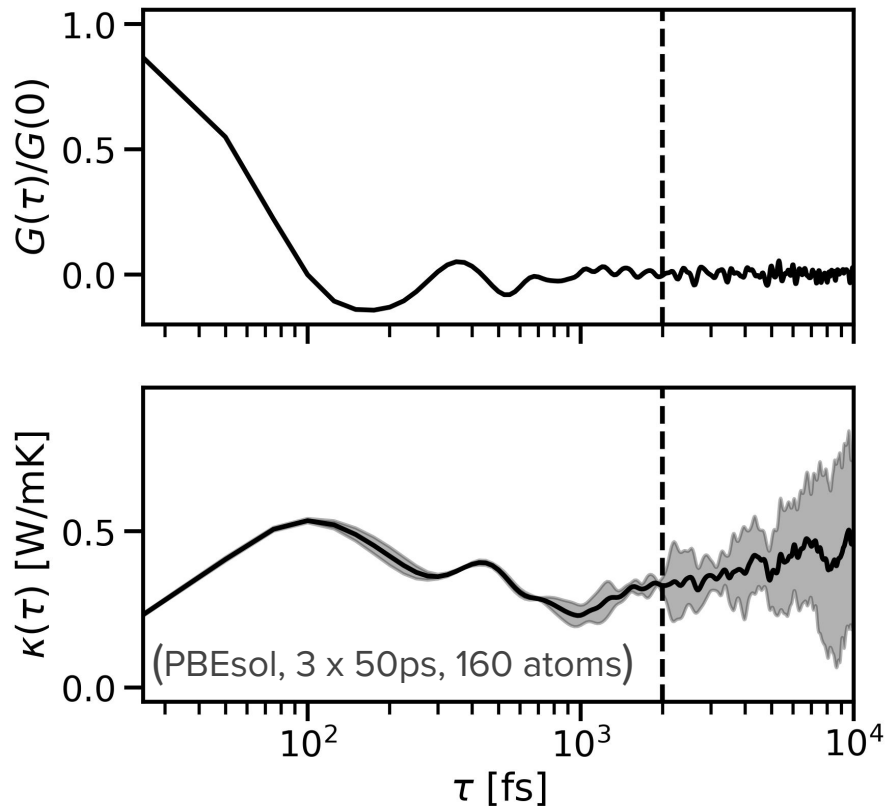
**Heat Flux  
Autocorrelation  
Function**



**No approximations to potential energy surface**

C. Carbogno, R. Ramprasad, M. Scheffler, *PRL* **118**, 175901 (2017)

# Results for CsPbBr<sub>3</sub> (300K)



CsPbBr<sub>3</sub> (*Pnma* 62, 20 atoms in unit cell):

$$\kappa = \mathbf{0.32 \pm 0.06 \text{ W/mK}}$$

(exp [1]: **0.42 ± 0.04 W/mK**)

[1] Lee et al., *PNAS*, **114**, 8693 (2017)

# How about Boltzmann Transport?

$$\kappa(T) = \frac{1}{V} \sum_q c_{q,T} \mathbf{v}_q^2 \tau_{q,T}$$

$$\mathcal{V}_{\text{pot}} (\{\mathbf{R}^0 + \mathbf{u}\}) = \frac{1}{2} \Phi \cdot \mathbf{u}^2$$

# How about Boltzmann Transport?

Harmonic approximation

$$\kappa(T) = \frac{1}{V} \sum_q c_{q,T} v_q^2 \tau_{q,T}$$

$$\mathcal{V}_{\text{pot}}(\{\mathbf{R}^0 + \mathbf{u}\}) = \frac{1}{2} \Phi \cdot \mathbf{u}^2$$

$$\Phi : D(\mathbf{q}) |\mathbf{q}\rangle = \omega_q^2 |\mathbf{q}\rangle \implies c_{q,T} = \frac{\partial E(T)}{\partial T} ; \quad v_q = \nabla_q \omega_q$$

# How about Boltzmann Transport?

$$\kappa(T) = \frac{1}{V} \sum_{\mathbf{q}} c_{\mathbf{q},T} v_{\mathbf{q}}^2 \tau_{\mathbf{q},T}$$

**Perturbative  
Treatment of  
Third Order Terms**

$$\mathcal{V}_{\text{pot}}(\{\mathbf{R}^0 + \mathbf{u}\}) = \frac{1}{2} \Phi \cdot \mathbf{u}^2 + \frac{1}{3!} \Psi \cdot \mathbf{u}^3$$

$$\Phi : \quad D(\mathbf{q}) |\mathbf{q}\rangle = \omega_{\mathbf{q}}^2 |\mathbf{q}\rangle \implies c_{\mathbf{q},T} = \frac{\partial E(T)}{\partial T} ; \quad v_{\mathbf{q}} = \nabla_{\mathbf{q}} \omega_{\mathbf{q}}$$

$$\Psi : \quad \Sigma(\mathbf{q}, T) \implies \tau_{\mathbf{q},T} = 2 / \text{Im } \Sigma(\mathbf{q}, T)$$

# How about Boltzmann Transport?

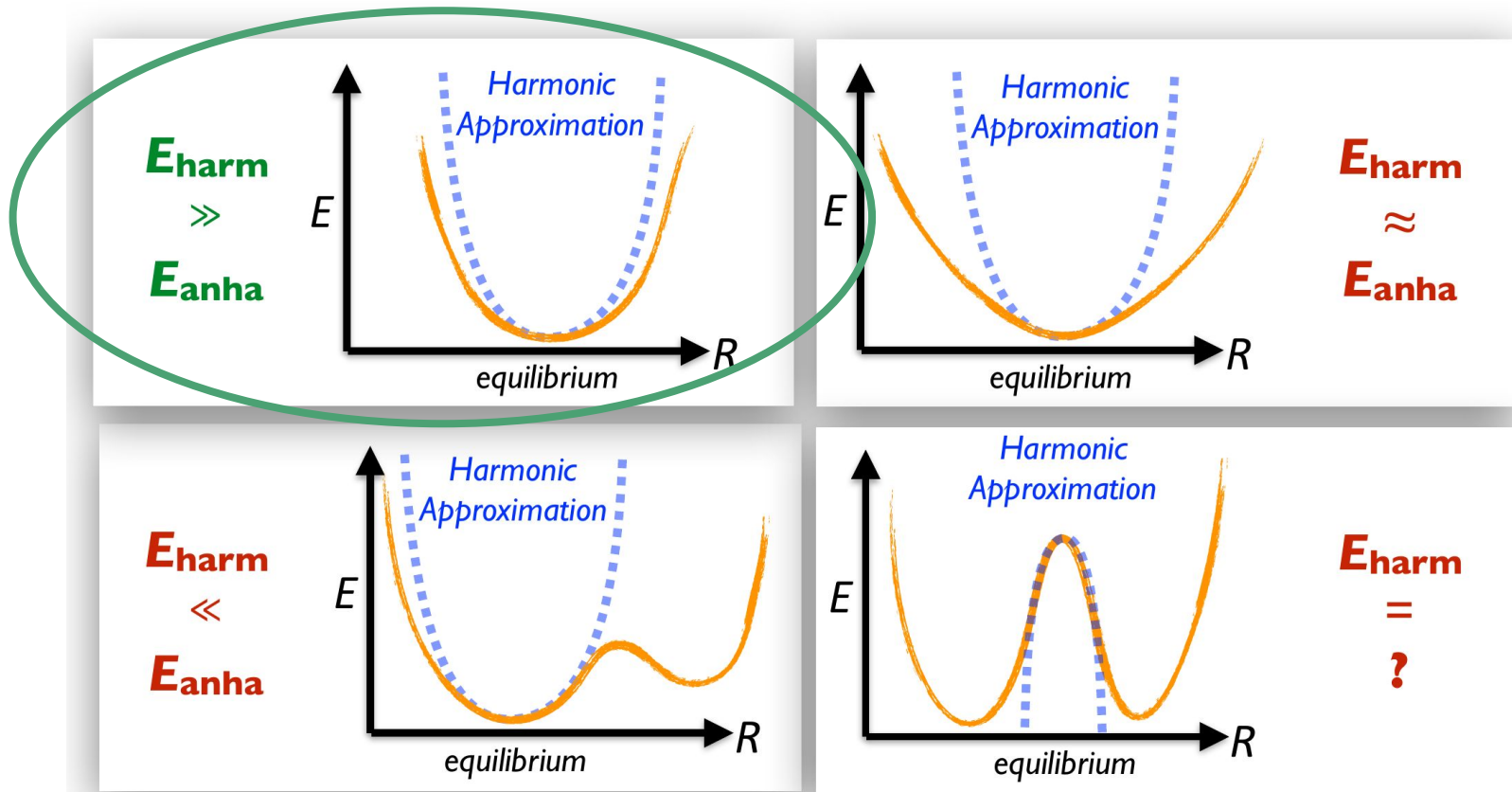
$$\kappa(T) = \frac{1}{V} \sum c_{\mathbf{q},T} \mathbf{v}_{\mathbf{q}}^2 \tau_{\mathbf{q},T}$$

Problem: Limits of perturbation theory?

$$\Phi : D(\mathbf{q}) |\mathbf{q}\rangle = \omega_{\mathbf{q}} |\mathbf{q}\rangle \implies c_{\mathbf{q},T} = \frac{\partial \omega_{\mathbf{q}}}{\partial T} ; \quad \mathbf{v}_{\mathbf{q}} = \nabla_{\mathbf{q}} \omega_{\mathbf{q}}$$

$$\Psi : \Sigma(\mathbf{q}, T) \implies \tau_{\mathbf{q},T} = 2 / \text{Im} \Sigma(\mathbf{q}, T)$$

# Why could perturbative treatment be insufficient?



Can we say something smart  
about the nature of anharmonicity  
before submitting a million core hours?

**Yes we can!**



# Anharmonicity -- Potential Energy Surface


Nuclear Hamiltonian in adiabatic approximation:

$$\mathcal{H}(\mathbf{P}, \mathbf{R}) = \mathcal{K}(\mathbf{P}) + \mathcal{V}(\mathbf{R})$$

Split potential into **harmonic** and **anharmonic contribution**:

$$\mathcal{V}(\mathbf{R}) = \mathcal{V}^{(2)} + \mathcal{V}'$$

 **Anharmonic Contribution!**

**phonopy** 

$$\mathcal{V}^{(2)}(\mathbf{R}) = \frac{1}{2} \sum_{\substack{I,J \\ \alpha,\beta}} \Phi_{\alpha,\beta}^{I,J} \Delta R_I^\alpha \Delta R_J^\beta$$

# Anharmonicity -- All about that force

$$\begin{aligned} \mathcal{V}(\mathbf{R}) &= \mathcal{V}^{(2)} + \mathcal{V}' \\ \mathbf{F}_I &= -\nabla_I \mathcal{V}(\mathbf{R}) \\ \mathbf{F}_I &= \mathbf{F}_I^{(2)} + \mathbf{F}_I' \\ \Rightarrow \mathbf{F}_I' &= \mathbf{F}_I - \mathbf{F}_I^{(2)} \end{aligned}$$

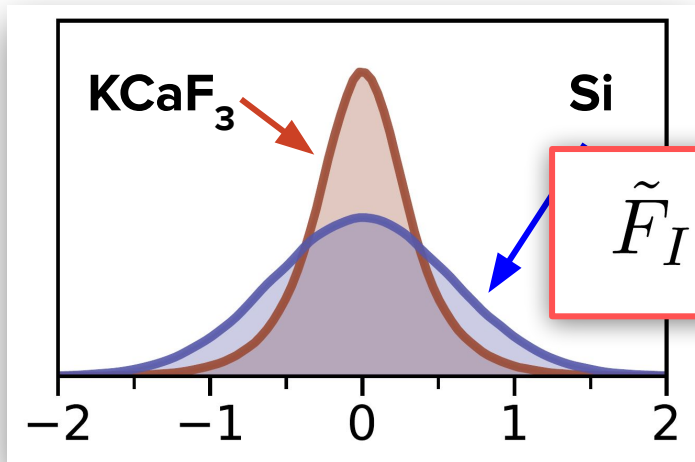
Anharmonicity of the potential  $\Leftrightarrow$  **anharmonic contribution to atomic forces**

# Conclusion

Maybe it is a clever idea to **look at anharmonic forces**.

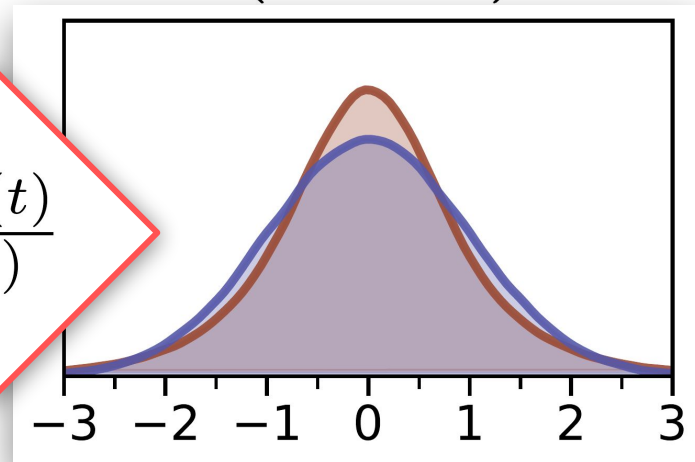
# How to compare forces?

$$p(F_{I,\alpha}(t))$$



$$\tilde{F}_{I,\alpha}(t) = \frac{F_{I,\alpha}(t)}{\sigma(\mathbf{F})}$$

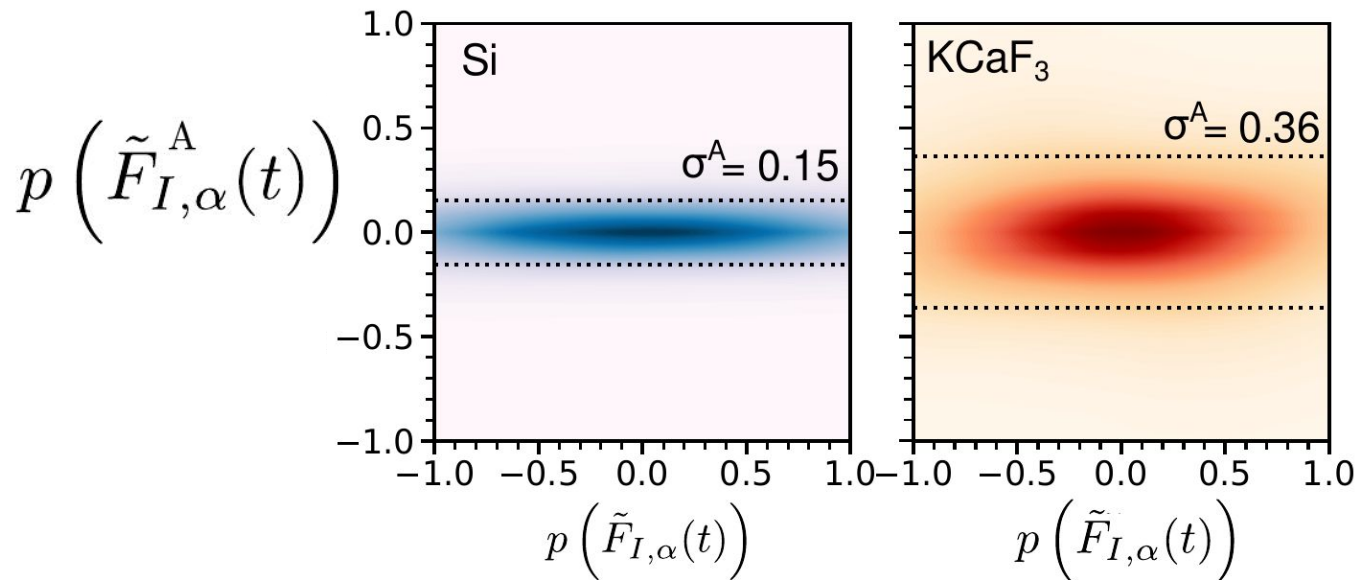
$$p(\tilde{F}_{I,\alpha}(t))$$



Sum up distributions of forces in each degree of freedom (atom, direction)  
 $\Rightarrow$  Variances add up

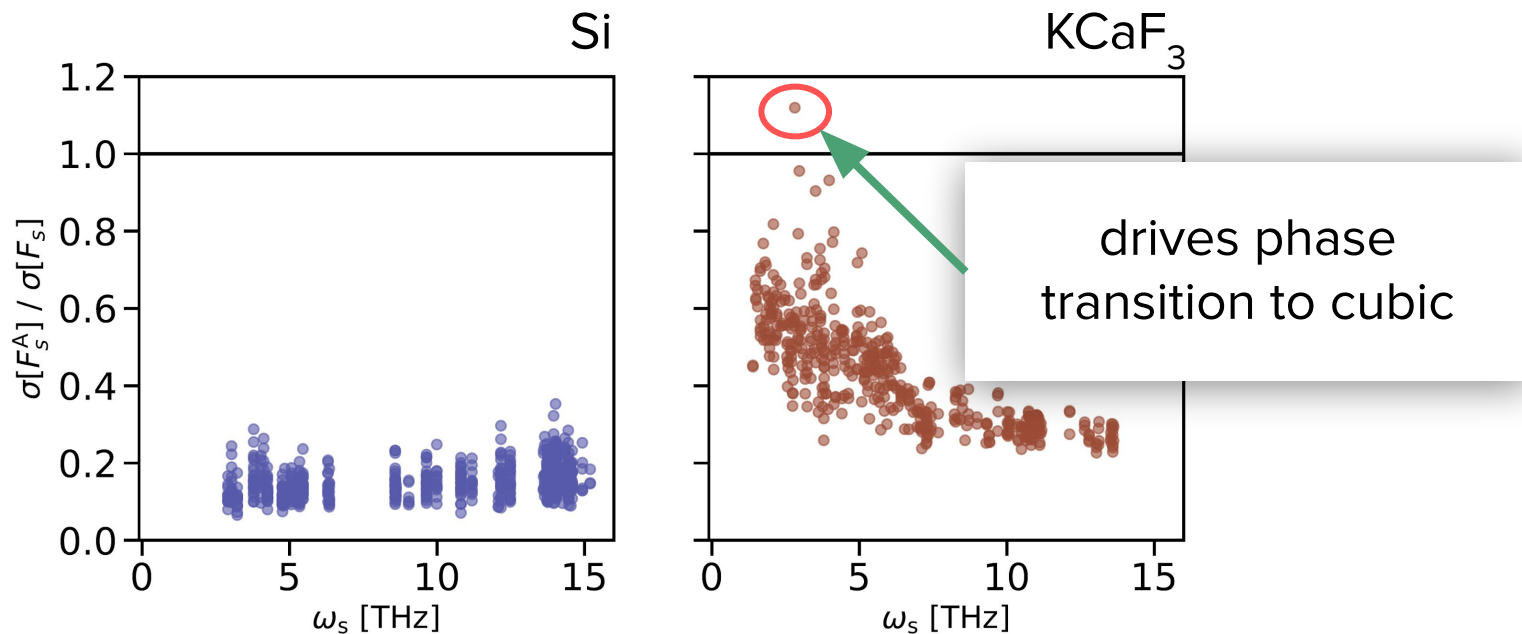
Forces observed during *ab initio* Molecular Dynamics runs

# Now: Measure the Scaled Anharmonic Force



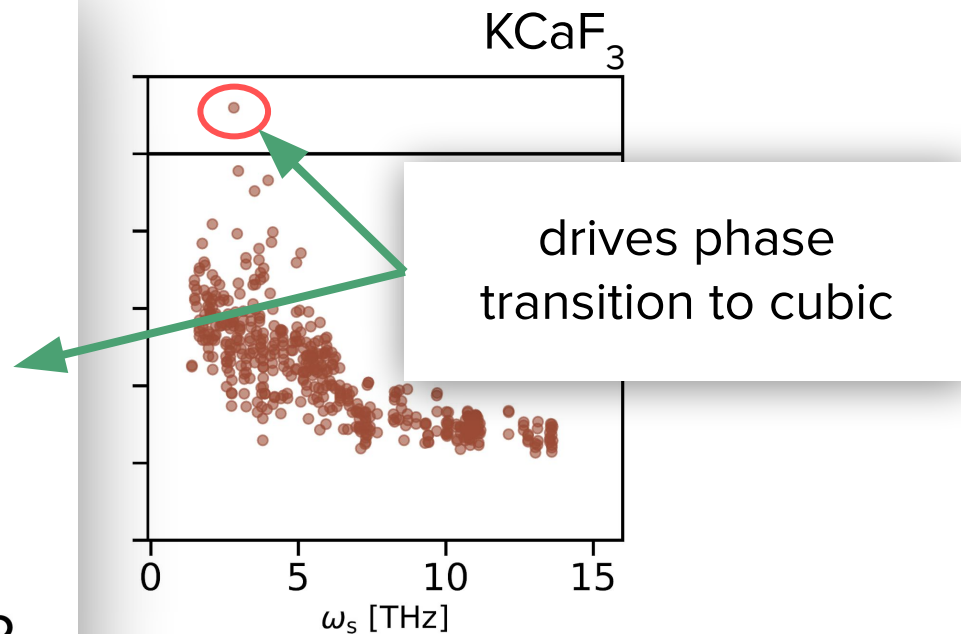
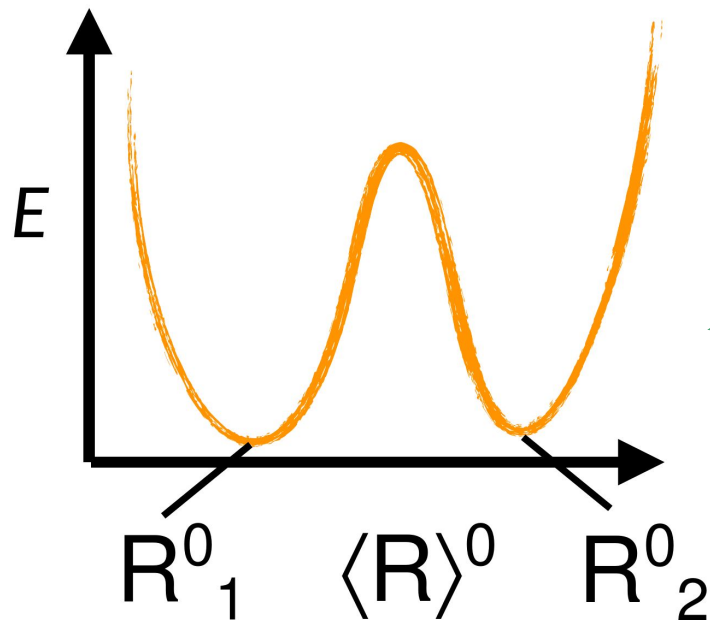
Distribution of anharmonic force contribution differs between compounds

# Project on Modes



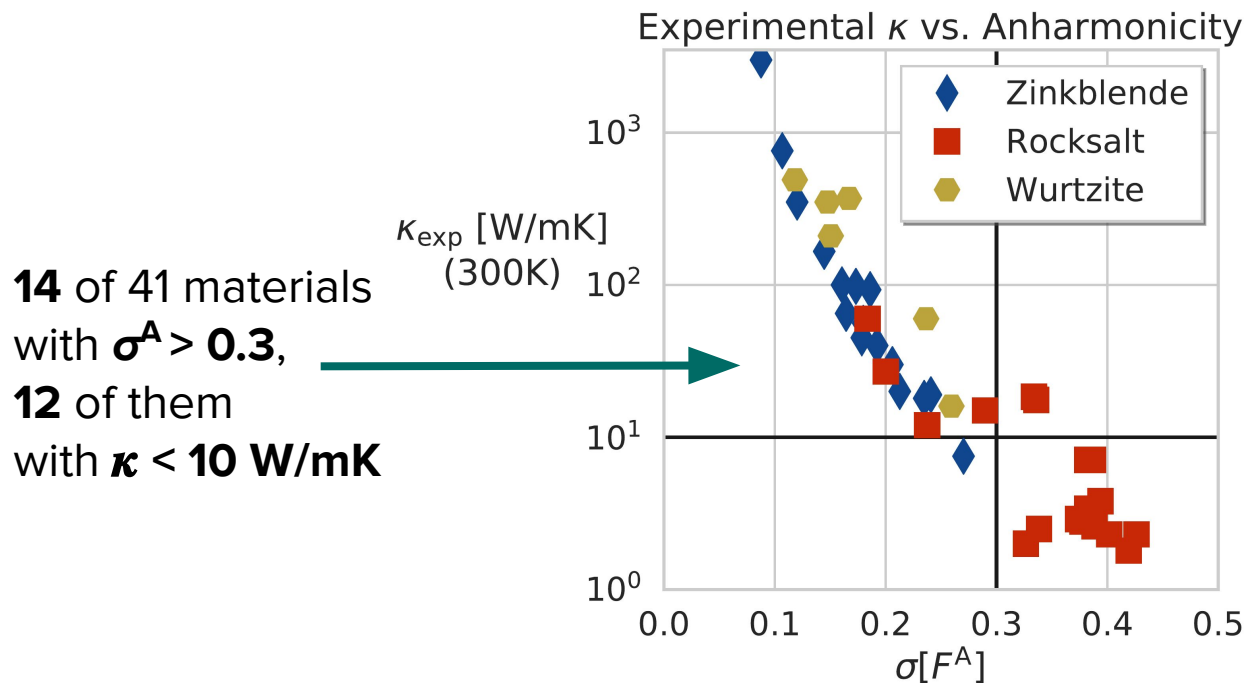
$$\mathbf{F}_s = \sum_I \frac{1}{\sqrt{M_I}} \mathbf{e}_{sI} \cdot \mathbf{F}_I$$

# Project on Modes



$$F_s = \sum_I \frac{1}{\sqrt{M_I}} \mathbf{e}_{sI} \cdot \mathbf{F}_I$$

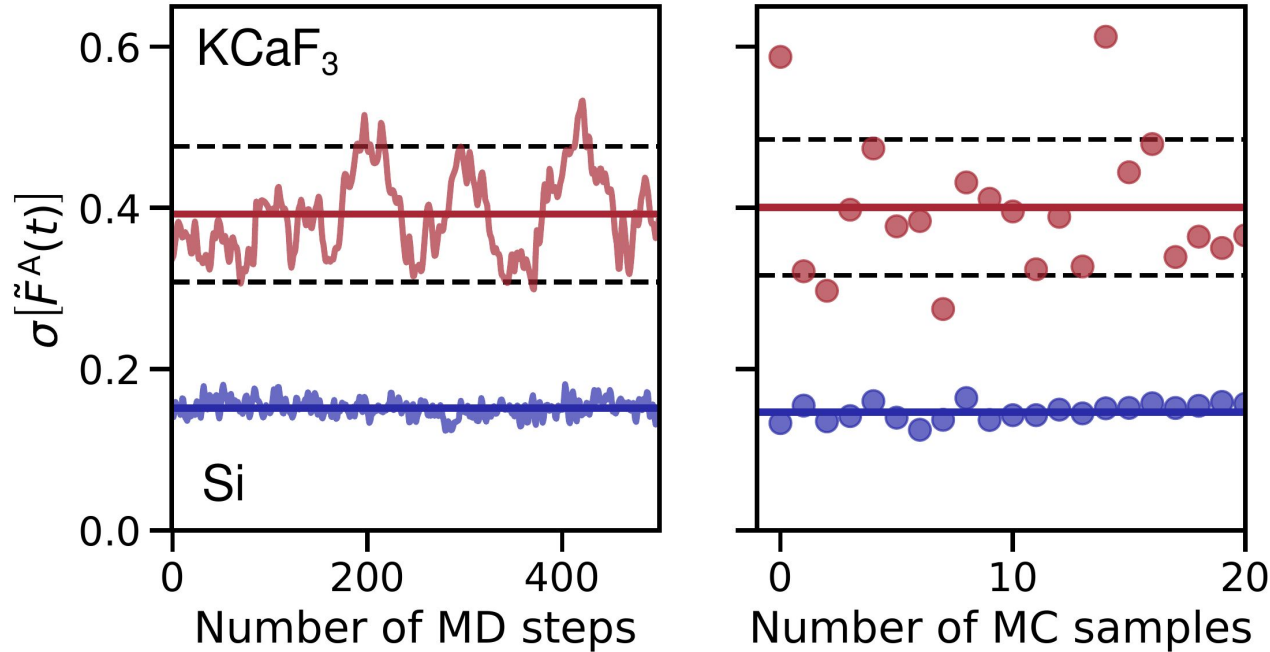
# Anharmonicity and Thermal Conductivity



“The more anharmonic a material the lower the thermal conductivity”  
is actually true!



# Can this be done efficiently? Yes.

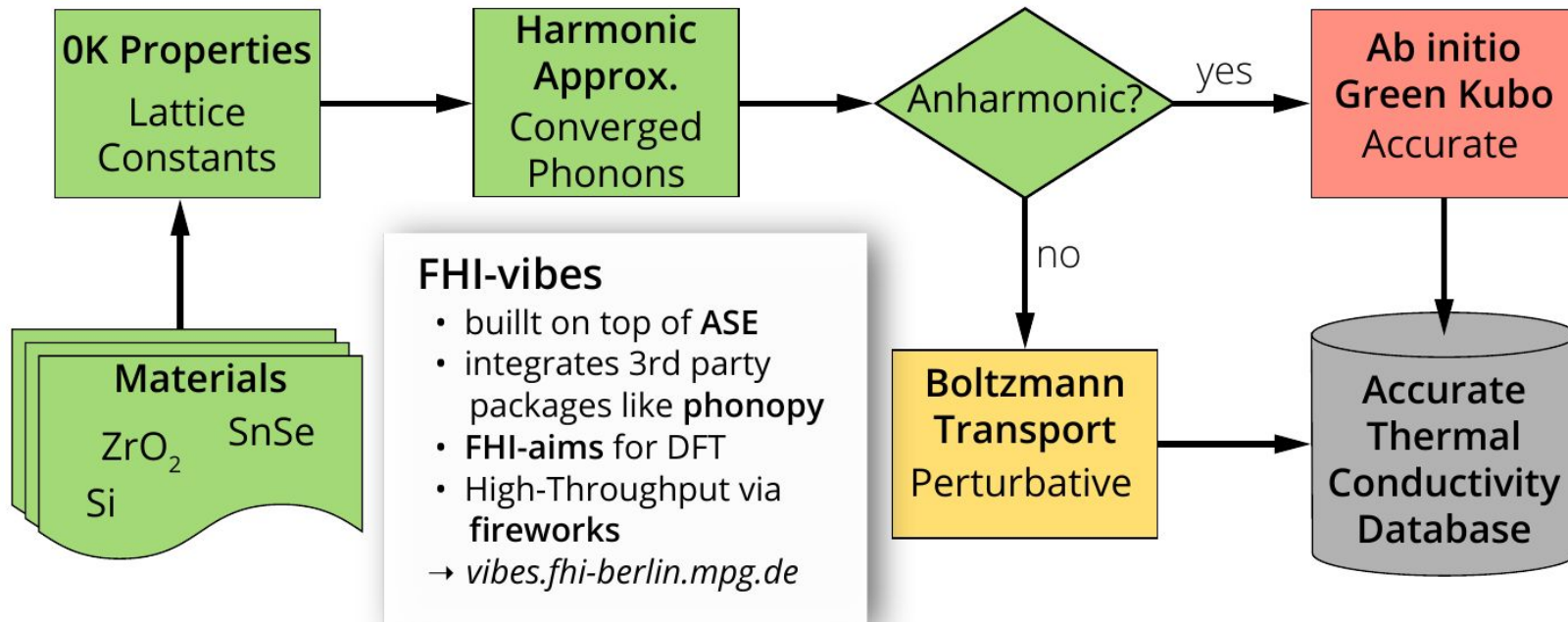


Molecular Dynamics  $\rightarrow$  Monte Carlo Sampling [1, 2]

[1] D. West and S.K. Estreicher, *Phys. Rev. Lett.* **96**, 115504 (2006)

[2] M. Zacharias and F. Giustino. *Phys. Rev. B* **94**, 175901 (2017)

# Our High-Throughput Approach for $\kappa$



# What we want from our framework

- ✓ Use existing methodologies (**phonopy**, `ase.md`)
  - **ase** makes it easy to combine functionality from different packages
- ✓ Efficient as possible DFT calculations
  - via `SocketIOCalculator`` (ase **3.17**)
- x Defined input and output files for **defined workflows**
  - simple input files via python configparser
  - **self contained** output files via our own output file format
- x Command line interface
- x Cluster support (slurm submission, restarts)
- x High-Throughput functionality
  - via FireWorks, not discussed today



**hilde** 🔒

Project ID: 8260678



☆ Star

3

🔗 Fork

1

Clone ▾

📄 ISC License   🔑 1,550 Commits   🌿 8 Branches   🏷 3 Tags   📁 21.3 MB Files

Lattice Dynamics with python

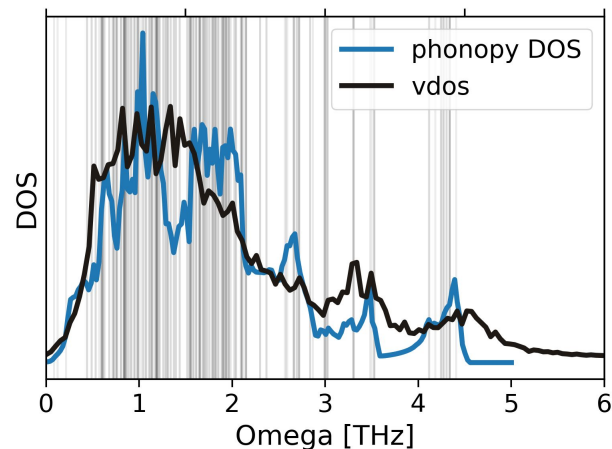
python 3.7+

code style black

pipeline passed

license ISC

- **Fritz Haber Institute vibrational analysis toolkit**
  - python package for **lattice dynamics** investigations with **FHlaims**
  - built on top of **ase**
- **Current Features**
  - **phonopy** calculations → Tutorial on Friday
  - **MD + heat flux** simulations
  - **toolset** for preparing, and analyzing data
  - high throughput functionality via **fireworks**
    - Tom Purcell



# Input File for MD: **md.in**

```
[geometry]
file:          geometry.in
primitive:     geometry.in.primitive
supercell:     geometry.in.supercell

[control]
xc:            pbesol
k_grid:       [2, 2, 2]
charge_mix_param: 0.3

[basissets]
default:      light

[md]
timestep:     5
maxsteps:     2000
driver:       Langevin
temperature:  300
friction:     0.02
compute_stresses: 10
```

# Input File for MD: **md.in**

```
[...]  
[slurm]  
nodes:                5  
  
[restart]  
command:              vibes submit md
```

for running  
on the cluster



# submit a job

```
➤ vibes submit md
```

```
Submitted batch job 10294567
```

# Output File: **trajectory.son**

```
{<<metadata json block>>}  
===  
{<<structure  0 data>>}  
---  
{<<structure  1 data>>}  
---  
{<<structure  2 data>>}  
---  
{<<structure  3 data>>}  
---  
...
```



# Output File: **trajectory.son**

```
{ "MD": {
  "type": "molecular-dynamics",
  ...},
"calculator": {
  "calculator": "Aims",
  "calculator_parameters": {
    "xc": "pbso1",
    "k_grid": [2, 2, 2],
    ...}},
"atoms": {
  "cell":
    [[ 8.33141234000000e+00, -8.33141234000000e+00,  0.00000000000000e+00],
     [ 8.33141234000000e+00,  8.33141234000000e+00,  0.00000000000000e+00],
     [ 0.00000000000000e+00,  0.00000000000000e+00,  1.24971185100000e+01]],
  "positions":
    [[ 0.00000000000000e+00,  0.00000000000000e+00,  0.00000000000000e+00],
     [ 4.16570617000000e+00,  0.00000000000000e+00,  0.00000000000000e+00],
     ...}
"primitive": { ...
...}
===
...
```

# Output File: **trajectory.son**

```
{<<metadata json block>>}  
===  
{  
  "atoms": {  
    "info": {  
      "nsteps": 0,  
      "dt": 4.91134739423203e-01,  
      "aims_uuid": "D985353A-F8FD-4635-A939-E129A7E6E146"},  
    "positions":  
      [[ 0.00000000000000e+00, 0.00000000000000e+00, 0.00000000000000e+00],  
       [ 4.16570617000000e+00, 0.00000000000000e+00, 0.00000000000000e+00], ...]  
    }  
  }  
}  
---  
{  
  "atoms": {  
    "info": {  
      "nsteps": 1,  
      "dt": 4.91134739423203e-01,  
      "aims_uuid": "D985353A-F8FD-4635-A939-E129A7E6E146"},  
    "positions":  
      [[ 1.62463733205010e-04, 7.52668360843155e-04, -4.84513363592926e-05],  
       [ 4.16666097188355e+00, -4.45877229739784e-04, -3.12871862455504e-04], ...]  
    }  
  }  
}  
---  
...
```

# Extract data from trajectory.son

> **vibes output md trajectory.son**

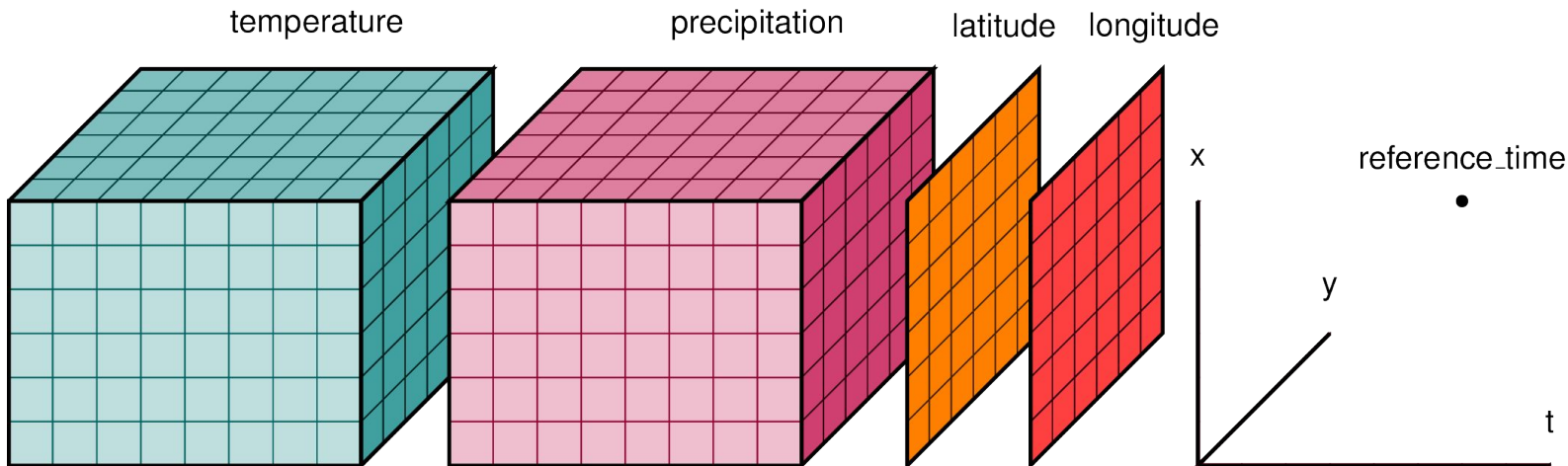
```
Extract Trajectory dataset from trajectory.son
[trajectory] Parse trajectory
[son] read file: trajectory.son
[son] process: ||||| 2002/2002
.. create atoms: ||||| 2001/2001
[trajectory] .. done in 1.639s
[trajectory] Get positions from trajectory
[trajectory] .. time elapsed: 0.038s
[trajectory] Get velocities from trajectory
[trajectory] .. time elapsed: 0.009s
[trajectory] Get pressure from trajectory
[trajectory] .. time elapsed: 0.308s
```

**Trajectory dataset written to trajectory.nc**

**trajectory.nc** : netCDF (hdf5) file containing an **xarray.Dataset**

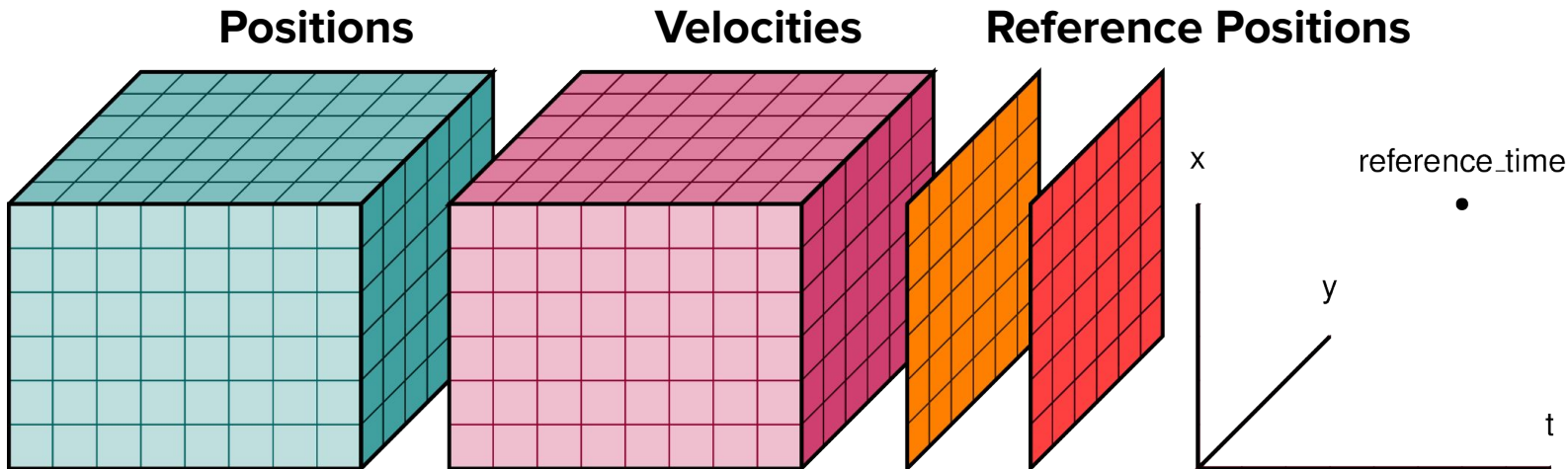
# ¿ xarray.Dataset ?

- Similar to [pandas.DataFrame](#)
  - for **multi-dimensional data** (former pandas.Panel)
- Philosophy: Data should be **bundled** and **labelled**
  - compare **.xyz files**
- [xarray.pydata.org/en/stable/why-xarray.html](http://xarray.pydata.org/en/stable/why-xarray.html)



# ¿ xarray.Dataset ?

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# FHI-vibes: inspect trajectory dataset

```
> vibes info netcdf trajectory.nc
```

```
<xarray.Dataset>
Dimensions:                (I: 120, a: 3, b: 3, time: 2001)
Coordinates:
  * time                    (time) float64 0.0 5.0 10.0 ... 9.99e+03 9.995e+03 1e+04
Dimensions without coordinates: I, a, b
Data variables:
  displacements            (time, I, a) float64 ...
  velocities               (time, I, a) float64 ...
  forces                   (time, I, a) float64 ...
  kinetic_energy           (time) float64 ...
  potential_energy         (time) float64 ...
  stress                   (time, a, b) float64 ...
  temperature              (time) float64 ...
  [...]
Attributes:
  System Name:             BaHfO3
  time unit:               fs
  timestep:                5.000000000000001
  nsteps:                  2000
  reference atoms:         {"cell": \n[[ 8.33141234000000e+00, -8.331412...
  average atoms:           {"cell": \n[[ 8.33141234000000e+00, -8.331412...
  reference positions:     [ 0.          0.          0.          4.16570...
  reference primitive atoms: {"cell": \n[[ 4.16570617000000e+00, 0.000000...
  [...]
```

# FHI-vibes: inspect trajectory dataset

```
> vibes info md trajectory.nc --plot
```

Dataset summary for trajectory.nc:

```
[info]          Summarize Temperature
```

```
Simulation time:      10.000 ps (2001 steps)
Temperature:          279.270 +/-      40.4779 K
Temperature (1st 1/3): 261.954 +/-      59.1326 K
Temperature (2st 1/3): 294.874 +/-      21.3062 K
Temperature (3st 1/3): 280.982 +/-      20.5579 K
```

```
[info]          Summarize Potential Energy
```

```
Simulation time:      10.000 ps (2001 steps)
Pot. Energy:          -15676395.140 +/-      0.6166 eV
Pot. Energy (1st 1/3): -15676395.381 +/-      0.9148 eV
Pot. Energy (2st 1/3): -15676394.917 +/-      0.3248 eV
Pot. Energy (3st 1/3): -15676395.121 +/-      0.3012 eV
```

```
[info]          Summarize Pressure
```

```
Simulation time:      10.000 ps (218 of 2001 steps)
Pressure:              0.792685 +/-      0.164142 GPa
Pressure (last 1/2):   0.792882 +/-      0.105375 GPa
Pressure (last 1/2):   0.004949 +/-      0.000658 eV/AA**3
```

```
[...]
```

```
.. summary plotted to md_summary.pdf
```

# FHI-vib

> vibes info md

Dataset summary

[info]

Simulation time:

Temperature:

Temperature (1st

Temperature (2st

Temperature (3st

[info]

Simulation time:

Pot. Energy:

Pot. Energy (1st

Pot. Energy (2st

Pot. Energy (3st

[info]

Simulation time:

Pressure:

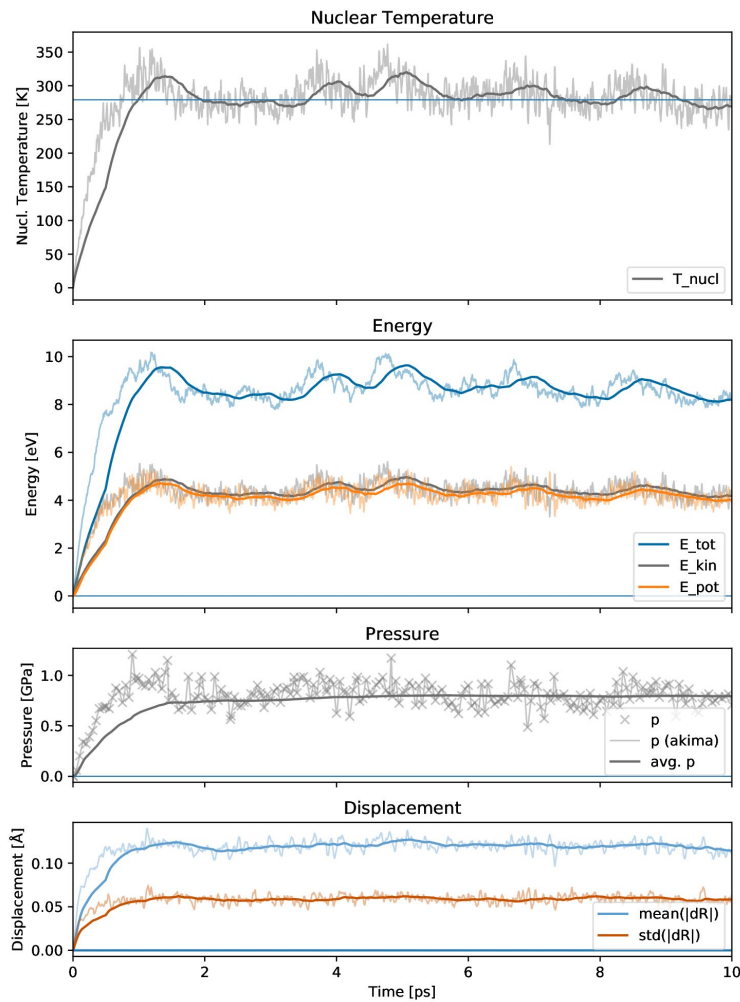
Pressure (last 1

Pressure (last 1

[...]

.. summary plott

dataset





# FHI-vibes: Outlook

- Poster Session: Let's discuss there!
- Tutorial on Friday:
  - use FHI-vibes to perform phonopy calculations
  - inspect an MD trajectory obtained with FHI-vibes and perform postprocessing
  - interoperability with hiphive
- Further Features:
  - high-throughput functionality via Fireworks developed by Tom Purcell
- hopefully released by the end of the year
- right now: invitation to our gitlab possible

**Thank you!**