

Design and characterization of a conductive nanostructured polypyrrole-polycaprolactone coated magnesium/PLGA composite for tissue engineering scaffolds

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Abstract: A novel biodegradable and conductive composite consisting of magnesium (Mg), polypyrrole-*block*-polycaprolactone (PPy-PCL), and poly(lactic-co-glycolic acid) (PLGA) is synthesized in a core-shell-skeleton manner for tissue engineering applications. Mg particles in the composite are first coated with a conductive nanostructured PPy-PCL layer for corrosion resistance via the UV-induced photopolymerization method. PLGA matrix is then added to tailor the biodegradability of the resultant composite. Composites with different composition ratios are examined through experiments, and their material properties are characterized. The *in vitro* experiments on culture of 293FT-GFP cells show that the composites are suitable for cell growth and culture. Biode-

gradability of the composite is also evaluated. By adding PLGA matrix to the composite, the degrading time of the composite can last for more than eight weeks, hence providing a longer period for tissue formation as compared to Mg composites or alloys. The findings of this research will offer a new opportunity to utilize a conductive, nanostructured-coated Mg/PLGA composite as the scaffold material for implants and tissue regeneration. © 2015 Wiley Periodicals, Inc. *J Biomed Mater Res Part A*: 103A: 2966–2973, 2015.

Key Words: conductive scaffold, tissue engineering, PPy-PCL, magnesium, nanostructured coating

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INTRODUCTION

Artificial tissue regeneration has emerged as a promising approach to repairing damaged or diseased tissues in human body. To produce three-dimensional (3D) tissue construction, the fabrication process usually requires a polymeric scaffold as a temporary platform for cells to adhere and proliferate. Hence, the polymeric scaffold material needs to be biocompatible and biodegradable such that cell growth, proliferation, and differentiation can be allowed.^{1,2} In addition, the scaffold material should be stiff enough for practical handling.³ Recent studies have shown that human tissues possess different dielectric properties^{4–6} and the presence of electrical signal can play an integral role in cell behavior and tissue formation.^{7–9} For instance, electrical stimulation has been found to be effective to enhance cell adhesion,¹⁰ cell differentiation,^{11,12} wound healing,¹³ protein secretion,¹⁴ and DNA synthesis of electrically responsive cells.^{15–17} Our recent research also shows that electric field

can be utilized as the active cell seeding mechanism to manipulate a large amount of cells toward the scaffold body.¹⁸ Therefore, synthesizing a conductive biomaterial that can be used for the above electrical micro-environment would have great potential to accelerate the tissue culture and development processes.

As one of the most abundant minerals in the body, magnesium (Mg) is essential to many cellular biochemical reactions such as energy metabolism, protein synthesis, and maintenance of the electrical potential of cell membranes.¹⁹ Some studies have shown that the use of magnesium supplementation may help reduce blood pressure^{20,21} and facilitate bone regeneration.²² With excellent mechanical properties and biocompatibility in human body, magnesium has been increasingly used as the biomaterial for medical implant applications.²³ To date, a number of magnesium-based scaffolds have been reported by different research groups.^{24–27} Nevertheless, many of them mainly used

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magnesium fibers or magnesium sheet as the material, which are hard to be handled by conventional micro-fabrication techniques such as solid freeform fabrication (SFF) or 3D printing to construct complicated 3D structure that is necessary for cell culture. Some investigations were performed recently for producing a topologically ordered Mg-biopolymer hybrid 3D composite structure,^{28,29} but the formed 3D structure was found to have a higher corrosion rate due to polymer breakdown.

Magnesium is a known reactive metal that is vulnerable to oxidation and other chemical reactions. Magnesium can react with water to produce magnesium ions that lead to a pH increase in the cell culture medium. Its fast degradation rate also makes it difficult to meet the specific requirement for clinical applications.^{30–32} The key challenge for Mg-based medical implants lies in the control of material stability and degradation in a physiological environment. Surface modification via coating can be a viable solution, and examples of the coating materials include poly(lactic-co-glycolic acid) (PLGA),³³ poly caprolactone (PCL),³⁴ and different conductive polymers (CPs).^{35–37} Polypyrrole (PPy) is a well-studied CP for tissue engineering because it possesses good electrical conductivity, biocompatibility, and high electrical stability.^{38,39} Mg alloy protection through coating of PPy powders was examined.²⁴ A recent study demonstrated that a block copolymer composed of PPy and PCL exhibited better conductivity, biodegradability, and ability to support PC12 cell proliferation.⁴⁰ PPy itself is generally non-degradable with limited manufacturability of complex 3D structures. Composite materials have the advantages of tailoring the material properties by adjusting the ratio among the materials. Some investigations have been performed recently for producing conductive composites such as Mg-PPy,²⁴ PLGA-Mg,^{25,33} and PLGA-PPy.^{38,39}

This article reports a new approach to synthesizing a novel composite that consists of Mg particles, PPy-PCL, and PLGA in a core-shell-skeleton manner. Mg particles were chosen as the conductive cores. Conductive polymer PPy-PCL was polymerized to form a nanostructured shell that encapsulates Mg particles from corrosion and oxidization. PLGA was used as the skeleton of the copolymer composite to provide controllability on the degradation rate. The material properties of the composite were characterized. Cell culture experiments were conducted on such a new composite, and the results demonstrated that this new composite exhibited excellent biomaterial properties, electrical conductivity, and good mechanical property for tissue engineering scaffolds. Findings of this study will lead generation of a new biomaterial that can be both conductive and biodegradable for a broad range of tissue engineering applications.

MATERIALS AND METHODS

Materials

The following materials, purchased from Sigma-Aldrich, were used in this study: PPy-PCL (polypyrrole-*block*-poly (caprolactone)), biodegradable block copolymers containing *p*-toluenesulfonate as dopant, 0.3–0.7 wt % in nitrome-

TABLE I. Concentration of PPy-PCL, PLGA, and Mg in the Composites

| Composite | Composition | PPy solution (mL) | Mg (mg) | PLGA (mg) |
|-----------|-------------|-------------------|---------|-----------|
| A1 | 1Py10 Mg | 1 | 10 | |
| A2 | 1Py20 Mg | 1 | 20 | |
| A3 | 2Py10 Mg | 2 | 10 | |
| B1 | 1Py10 Mg4PG | 1 | 10 | 4 |
| B2 | 1Py20 Mg4PG | 1 | 20 | 4 |
| B3 | 2Py10 Mg8PG | 2 | 10 | 8 |

thane), Mg(Granular, 20–230 mesh, 98% Mg, reagent grade), poly(lactic-glycolic acid) (PLGA, ratio M/M% 50:50, MW 30,000–60,000Da), dichloromethane(Anhydrous), acetone ($\geq 99.9\%$), phosphate buffered saline (PBS) tablets, and 0.5M hydrochloric acid. All other chemicals were analytical grade and were used as received.

Preparation of nanostructured PPy-PCL coated Mg particles

The UV-induced polymerization method was used for coating PPy-PCL on the Mg particles. Prior to each experiment, Mg particles were immersed in 0.5 mol/L hydrochloric acid for 30 s to remove the oxide layer on the surface. The particles were rinsed and kept in acetone before use. The photopolymerization process was carried out by dispersing the Mg particles into the PPy-PCL solution under a fume hood. Based on the work,⁴¹ the mixture was exposed to a 254 nm UV lamp (500 W) and simultaneously stirred with a magnetic stirrer at 500 R.P.M. until dried. UV irradiation was used to enhance the polymerization of polypyrrole when depositing onto the Mg particles. The dried Mg particles were further exposed to the UV lamp for 24 h at room temperature (RT). The photopolymerized Mg particles were washed twice with ethanol and dried under vacuum. In this work, the composition ratio between the Mg particles and the PPy-PCL solution was examined in order to characterize their influence on the resultant Mg particles. Table I summarizes the parameters used in the experiment.

Preparation of nanostructured PPy-PCL coated mg and PLGA composite

To fabricate the PPy-PCL coated Mg-PLGA composite scaffolds, PLGA powders were first dissolved in dichloromethane (CH_2Cl_2) to prepare a 10% (weight/volume) PLGA stock solution. Then, PPy-PCL coated Mg particles were added into the PLGA solution. The mixture was placed under vacuum to allow the PLGA solution to infiltrate into the porous PPy-PCL coated Mg aggregates, resulting in the formation of the PPy-PCL coated Mg and PLGA composite. The composite mixture was then cast into a Teflon petri dish and sonicated for 10 min (W-385, Farmingdale). After sonication, the composite was dried in a fume hood for overnight. The composite was further dried in an oven at 45 °C for 6 h and stored at room temperature for at least 2

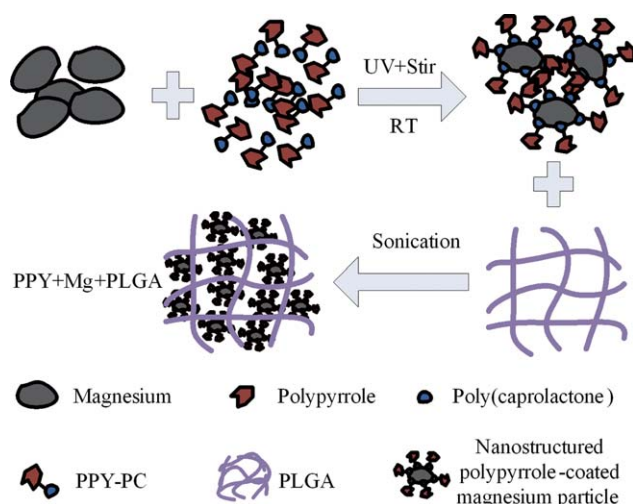


FIGURE 1. Schematic illustration for preparing the nanostructured PPY-PCL coated Mg particle and PLGA composite in a core-shell-skeleton manner. [Color figure can be viewed in the online issue, which is available at wileyonlinelibrary.com.]

days to evaporate all organic solvents. Figure 1 illustrates a schematic of the fabrication procedure for the nanostructured PPY-PCL coated Mg and PLGA composite. Composites with different composition ratios were prepared to evaluate their influence on the material properties and *in vitro* cell culture. Table I summarizes the amount of each material used during the photopolymerization and the composite synthesis processes. Prior to the experiments, each sample was treated with a plasma cleaner (PDC-002, 13.56 MHz RF Plasma) to enhance the surface hydrophilicity for cell attachment and cell growth.⁴² The pressure of the vacuum chamber was set to around 800 mTorr and the samples were exposed to the plasma from 45 W RF coil for 15 min.

The size and morphology of each composite were characterized using scanning electron microscopy (JSM-5600, JEOL Ltd). The accelerating voltage was set to 20 kV. To evaluate the repeatability of the results, three samples were tested per each composite group ($n = 3$), unless otherwise specified.

Mechanical testing

The compressive yield strength of each composite was measured using a micro hardness tester (Fischerscope HM 2000 XYp). The composites prepared in section "Preparation of nanostructured PPY-PCL coated Mg and PLGA composite" were first cut into pieces of 25 mm by 25 mm, with a thickness of ~ 0.8 mm. Before the testing, each sample piece was placed under a hydraulic press at 5 MPa for 10 min to enhance the surface uniformity. A test load of 100 mN was applied to the sample and the indentation depth was recorded. Based on the information, the unloading curve of each sample was plotted and the modulus of elasticity, E , was then calculated by Power Law Method. The modulus was averaged from the three samples of each group.

Conductivity measurement

The resistivity of the each composite was measured using the method as described.⁴³ Two metal electrodes that were

10 mm apart were first attached to a polydimethylsiloxane (PDMS) substrate and the electrodes were connected to a Digital Multimeter (2100/120, Keithley Instruments). Hydrated samples that are 10 mm long with a cross-sectional area of $0.8 \text{ mm} \times 0.8 \text{ mm}$ were tightly clamped between the two electrodes for the measurement. The resistance measurement (R) was carried out six times on each sample. The average conductivity, which is the inverse of the resistivity (ρ), was calculated using the Pouillet's law Eq. (1):

$$R = \frac{\rho L}{A} \quad (1)$$

where L is length and A is the cross-sectional area of the sample.

In vitro degradation measurement

In vitro degradation studies were performed by measuring the weight loss of the samples at different time during the incubation. All samples were sterilized with ethyl alcohol ($>99\%$) for 2 min, and rinsed twice with PBS before transferring to 12-well tissue culture plates for subsequent incubation. Dulbecco's Modified Eagle's Medium (DMEM) with 1% penicillin and streptomycin (Sigma-Aldrich) were added to each well to completely cover the sample surface and the plates were incubated at 37°C and 5% CO_2 for up to 8 weeks. At each predetermined interval, samples from each group were collected from the plate. The samples were washed with deionized water and then freeze-dried to measure the residual weight.

Cell viability and proliferation

A series of cell culture experiments were performed to examine the effects of different composition ratios on cell viability and proliferation. Sterilized samples were first transferred to a 24-well culture plate that was pre-coated with bovine serum albumin (BFA) to minimize cell attachment. Human kidney cells, 293FT, were stably transfected with a green fluorescent protein (GFP) plasmid and then pipetted onto the sterilized samples at a concentration of 5×10^4 cells. Following cell attachment at 37°C for 30 min, DMEM culture medium was added to the plate and the medium was exchanged every two to three days. To characterize the cells after culture, hydrated samples seeded with cells were first fixed with 4% paraformaldehyde (Sigma-Aldrich) for 20 min. The samples were then washed with PBS and dehydrated through an automated critical point dryer (CPD300, Leica Microsystems Ltd). The samples were sputter-coated with a 10 nm gold layer using a Quorum Coater (Q150TS, Quorum Technologies Inc.).

RESULTS

Microstructure, elemental composition, and size of PPY-PCL coatings

Material samples as prepared in the previous Section were characterized using scanning electron microscopy (SEM). Images obtained confirm that PPY-PCL was successfully coated onto the Mg particles via the photopolymerization

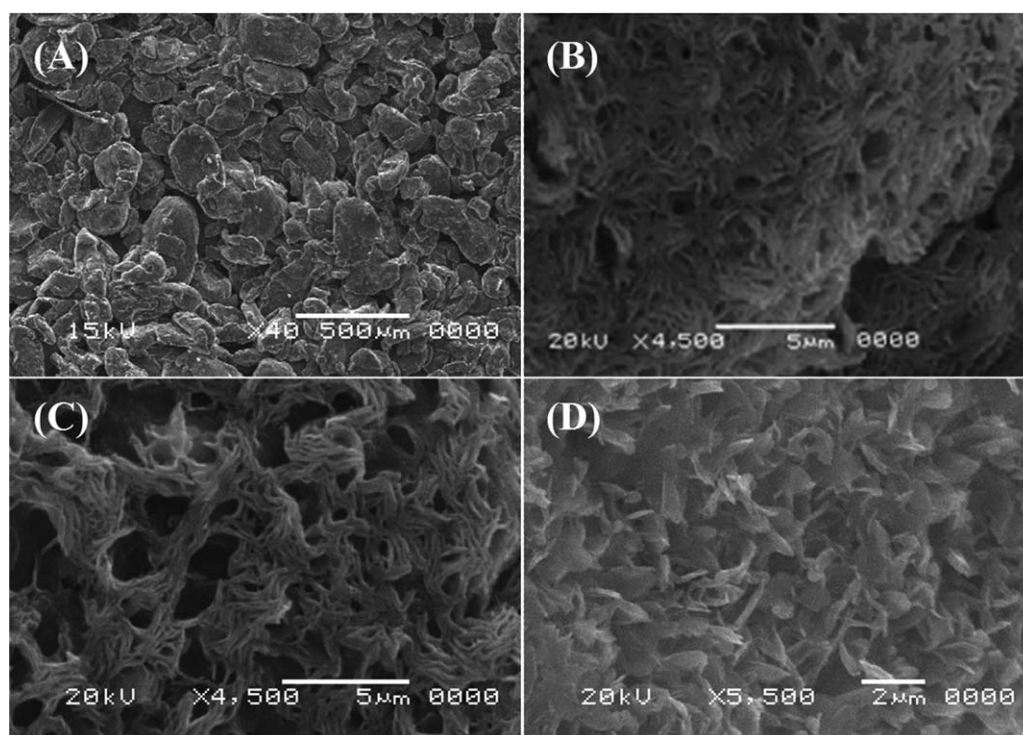


FIGURE 2. SEM images of nanostructured PPy-PCL coated Mg particles. A: 1Py10 Mg at low magnification (scale bar: 500 μm). B: 1Py10 Mg at high magnification (scale bar: 5 μm). C: 1Py20 Mg at high magnification (scale bar: 5 μm). D: 2Py10 Mg prepared at high magnification (scale bar: 2 μm).

method. Figure 2(A) shows the surface morphology of the material from PPy-PCL coated Mg particles (sample A1). Under a $\times 4500$ magnification, the PPy-PCL coating was found to have a relatively porous, nanoscale architecture [Fig. 2(B)].

Material properties

Table II summarizes the elastic modulus of each composite obtained from the mechanical compression testing. The densities were evaluated by measuring the dimensions and the weight of each sample. Among the three materials in the composite, Mg has the highest elastic modulus and the resultant modulus of the composite has a strong dependency on its ratio in the composite. For a typical bone tissue, the elastic modulus is around 0.12–0.5 GPa but the elastic modulus of PPy-PCL coated Mg is 18 ± 0.1 GPa. After adding PLGA into the composite, the average elastic modulus was reduced to 0.26 ± 0.01 GPa (composite B1), which was more favorable for tissue engineering application.

The conductivity of the three Mg-PPy-PLGA composites were also measured and shown in Table II. Composite B1 has a conductivity of 45 S/m. The conductivity of composite

B2 increased to 54 S/m due to a higher Mg content. Composite B3, which has the highest PLGA content, has a conductivity of only 41 S/m.

Degradation measurement

The degradation rate of the composite in DMEM medium was further examined. Figure 3 shows the change of the surface morphology of the composite B3 for 4 weeks. Before the experiment, the composite surface was mainly composed of PLGA matrix. After one week, cracks can hardly be seen on the PLGA surface, but more pores can be found. The PLGA matrix degraded faster after two weeks, and PPy-PCL coated Mg particles that were originally encapsulated by PLGA became exposed to the medium. After four weeks, PLGA started to lose its integrity and PLGA fragment were observed on the surface. Coated-Mg particles were also found to be corroded by the medium. The remaining weights of this composite B3, as well as the other two composites B1 and B2, were all measured and plotted in Figure 4. The results showed that the ratio of the PLGA matrix in the composite has a significant influence on the degradation behavior. After the first week, the percentages of average weight loss

TABLE II. Mechanical and Conductive Properties of Composites in Comparison With Spongy Bone

| Scaffold | Modulus (E) (GPa) | Density(ρ) (g/cm ³) | Specific modulus (E/ρ) | Conductivity (S/m) |
|-----------------------------|-----------------------|--|-------------------------------|--------------------|
| Spongy bone ^{5,42} | 0.12–0.5 | 0.14–1.2 | ~ 0.5 | 0.01–0.06 |
| 1Py10 Mg4PG (B1) | 0.26 ± 0.01 | 1.35 ± 0.03 | 0.19 | 45 ± 2 |
| 1Py20 Mg4PG (B2) | 0.38 ± 0.01 | 1.45 ± 0.04 | 0.26 | 54 ± 2 |
| 2Py10 Mg8PG (B3) | 0.31 ± 0.01 | 1.38 ± 0.05 | 0.23 | 41 ± 2 |

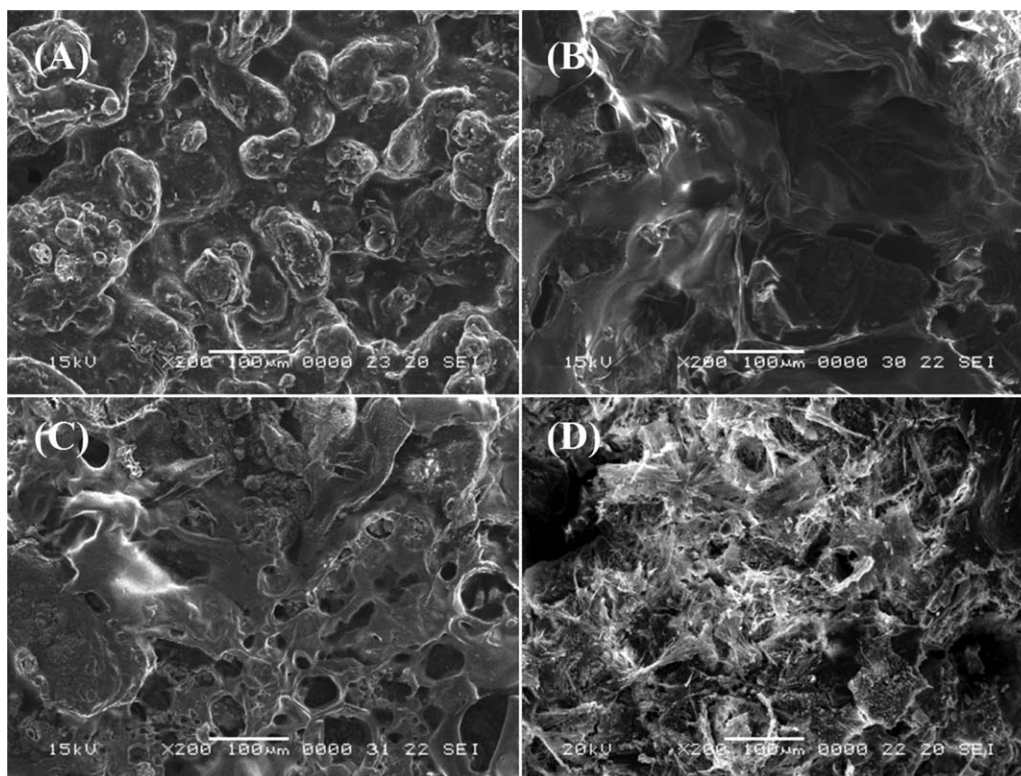


FIGURE 3. Surface morphology of the 2Py10 Mg8PG composite (B3) after immersing in DMEM solution. A: Zero. B: One week. C: Two weeks. D: Four weeks. Scale bar are 100 μ m.

for B1, B2, and B3 were 2%, 3%, and 1%, respectively. After two weeks, the degradation increased and the composite with low PLGA content and thinner coating (B2) has the highest degradation rate. The composite B3 still had at least 50% of the weight after four weeks. Another 0.7 mm thickness composite 1Py10 Mg (A1), without PLGA, was also fabricated and kept in DMEM as a control. This composite completely degraded in 4 weeks (9.6 mm/year). While the composites of PPy-PCL coated Mg with PLGA (B1, B2 and B3) were not completely degraded even after 8 weeks.

***In vitro* cell culture and proliferation**

The composites with PLGA covalently conjugated onto the nanostructured PPy-PCL coated Mg particles were examined *in vitro* for cell culture. Each composite seeded with 293FT-GFP cells was cultured for 5 days and the composites were observed under fluorescent microscopy (FM) and SEM every two days. The results (Fig. 5) show that 293FT-GFP cells were successfully adhered and proliferated throughout the composites. The coverage of the fluorescent green 293FT-GFP cells were quantified using image analysis software, and the data were plotted in Figure 6. Among the three composites (B1, B2, B3) with different composition ratios, B3 appears to be the most suitable substrate for cell culture and the cells proliferated at a rate of 4.6 times in 120 h. The composite B1 has a proliferation rate of 4.3 times while B2 has a rate of four times. This result can be explained by the change of pH value of the culture medium. A pH meter (SB70P, VWR International) was used to measure the pH values of the

post-culture media, and the results are shown in Figure 7. The pH values of B1 and B3 are close to that of the culture medium (control) in the incubator. In contrast, the pH value of B2 increases from 7.4 to 7.8, indicating a more alkaline solution due to the corrosion of the Mg particles.

DISCUSSIONS

Composition ratios affect the architecture and properties of material

The above results indicate that the composition ratio would have an influence on the architecture of the coating. As the

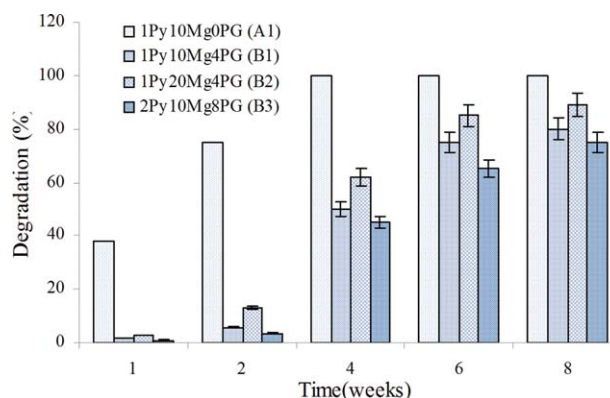


FIGURE 4. *In vitro* degradation of the control and the three composites: 1Py10 Mg4PG, 1Py20 Mg4PG, and 2Py10 Mg8PG (mean \pm standard deviation, $n = 3$) over eight weeks. [Color figure can be viewed in the online issue, which is available at wileyonlinelibrary.com.]

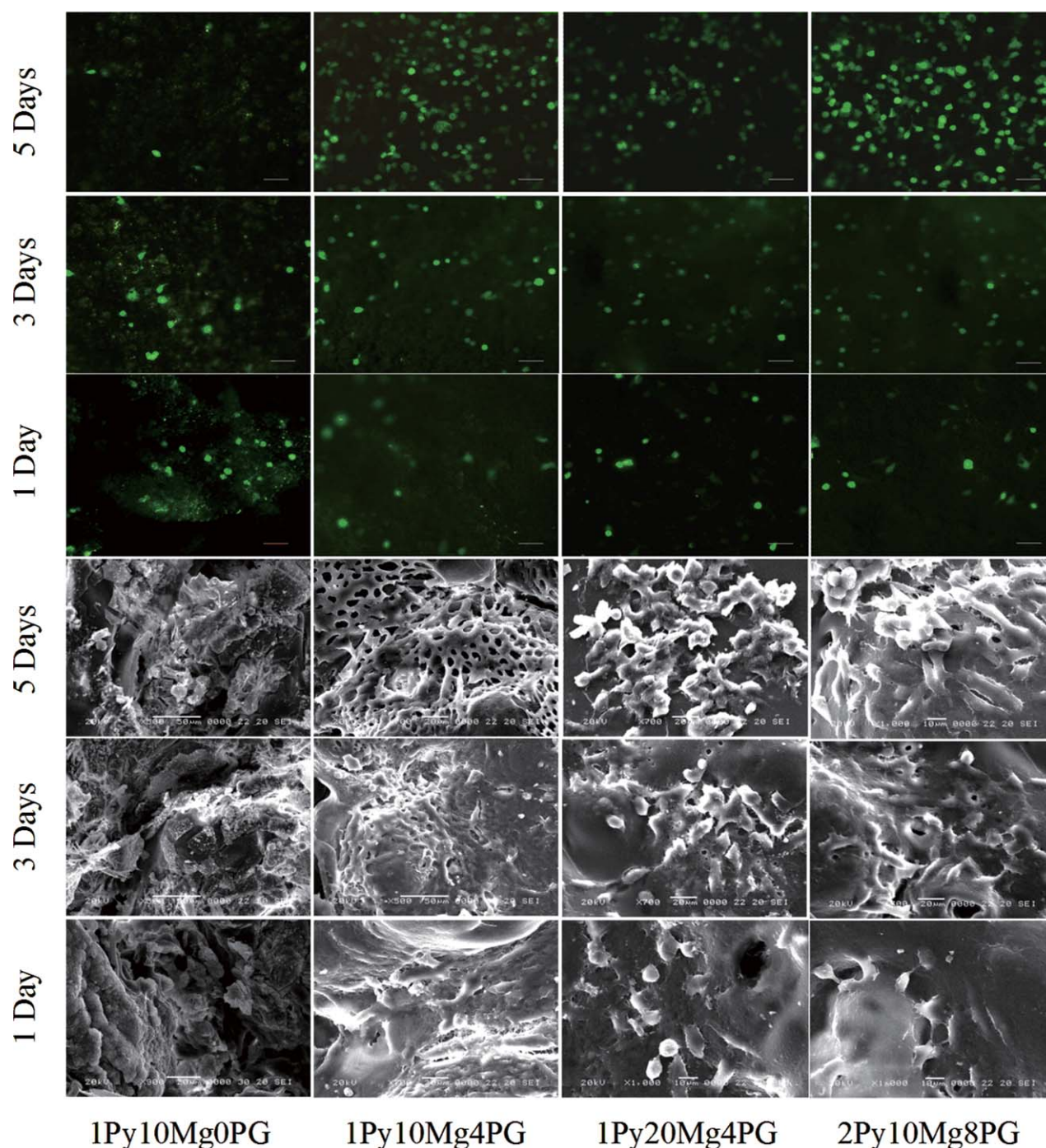


FIGURE 5. FM and SEM images of 293FT-GFP cells on different composite surfaces over the culture period (scale bar: 100 μm in FM, 10–50 μm in SEM). [Color figure can be viewed in the online issue, which is available at wileyonlinelibrary.com.]

content of the Mg particle increases, the coating becomes more porous and the average pore size changes from $\sim 2\ \mu\text{m}$ [sample A1, Fig. 2(B)] to $5\ \mu\text{m}$ [sample A2, Fig. 2(C)]. In contrast, increasing the content of the PPy-PCL solution leads to a denser surface coating [sample A3, Fig. 2(D)], and hence, pores are barely visible and the average size reduces to less than $0.5\ \mu\text{m}$.

Increasing the Mg ratio can also enhance the elastic modulus of the resultant composite, which was evidenced

by the increased modulus for composite B2, that is $0.38 \pm 0.01\ \text{GPa}$. The mechanical properties of all the three composites, 1Py10 Mg4PG (B1), 1Py20 Mg4PG (B2), and 2Py10 Mg8PG (B3), are comparable to a bone in human, which can provide sufficient mechanical strength to resist contractile forces after scaffold implantation.

The conductivity of the three Mg-PPy-PLGA composites indicates that the conductivity is dependent on the amount of the conductive PPy-PCL coated Mg filler in the composite.

In addition, all these composites are found to be more conductive than ventricular muscle, blood, and skeletal muscle in human (0.03 – 0.6 S/m),⁴⁴ demonstrating the feasibility of applying electrical signals or stimulations on these new biomaterials to accelerate the tissue regeneration.

Controlling mg degradation through PPy-PCL coatings better cell proliferation

The composite 1Py10 Mg (A1, without PLGA, 0.7 mm thickness sample) is completely degraded in four weeks (9.6 mm/year). While the composites of PPy-PCL coated Mg with PLGA (B1, B2, and B3) were not completely degraded even after 8 weeks. This result is comparable with the degradation rate of pure magnesium (10.5–39 mm/year), depending on the medium used and the material porosity.^{27,35} Therefore, PPy-PCL coated magnesium has better corrosion resistance than pure magnesium, and adding PLGA can enhance the stability and biodegradation resistance of the composite.

Mg particles in the composite react with the aqueous environment to form magnesium hydroxide. This hydrogenation causes the pH to increase and hence affects the cell culture.⁴⁵ This result agrees to the degradation study that the composite B2 has the fastest degradation rate among the three composites.

CONCLUSIONS

A biodegradable and conductive composite consisting of Mg, PPy-PCL, and PLGA was designed and fabricated. Three composites with different composition ratios were characterized for potential tissue engineering applications. Mechanical and biodegradability tests were conducted and the correlations among the three materials on the resultant composite were investigated. The results indicated that the use of Mg particles coated with conductive PPy-PCL can provide electrical conductivity to the resultant composite. Composites with high magnesium contents can improve the overall elastic modulus. Composites with high PLGA contents can extend the degradation time. *In vitro* experiments of culturing 293FT-GFP cells were conducted on the composites, and the cells were found to be proliferated and uni-

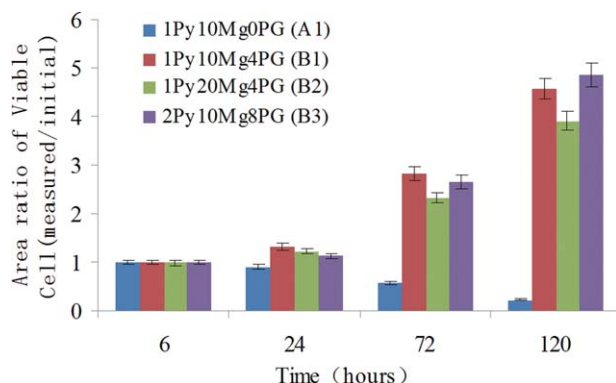


FIGURE 6. The results of cell proliferation. The area ratio covered by viable 293FT-GFP cells on the 1Py10 Mg0PG, 1Py10 Mg4PG, 1Py20 Mg4PG, and 2Py10 Mg8PG composites. [Color figure can be viewed in the online issue, which is available at wileyonlinelibrary.com.]

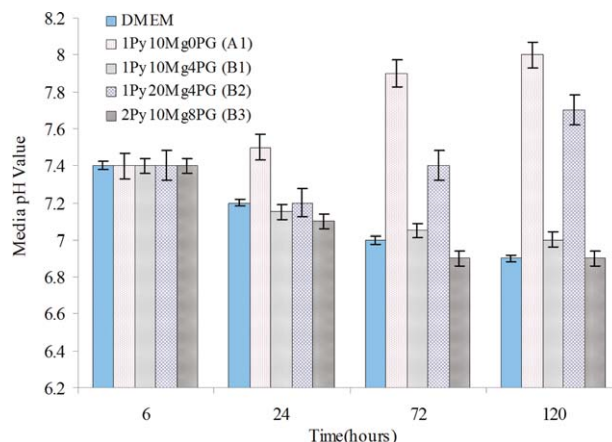


FIGURE 7. The pH values of the post-culture media. Average pH values of the post-culture media from the 1Py10 Mg0PG, 1Py10 Mg4PG, 1Py20 Mg4PG, and 2Py10 Mg8PG composites at different intervals. [Color figure can be viewed in the online issue, which is available at wileyonlinelibrary.com.]

formly spread throughout the composite surface. The results indicate that the newly designed biodegradable and conductive Mg-PPy-PLGA composites can be used for fabricating tissue engineering scaffolds, as well as other tissue engineering applications such as electrical stimuli and DEP cell pattern to enhance tissue regeneration.

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