Johannes Gutenberg Universität Mainz

DOCTORAL THESIS

Structure, Dynamics and Vibrational Spectroscopy of Interfacial Alkali Nitrate Aqueous Solutions from ab initio Molecular Dynamics

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A thesis submitted in fulfillment of the requirements for the degree of Doctor of Philosophy

in the

Condensed Matter Theory Group Institute for Physics

Declaration of Authorship

I, Gang Huang, declare that this thesis titled, "Structure, Dynamics and Vibrational Spectroscopy of Interfacial Alkali Nitrate Aqueous Solutions from *ab initio* Molecular Dynamics" and the work presented in it are my own. I confirm that:

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"If a problem is worth looking at at all, then no mathematical technique is to be judged too sophisticated."

David Ruelle

JOHANNES GUTENBERG UNIVERSITÄT MAINZ

Abstract

Condensed Matter Theory Institute for Physics

Doctor of Philosophy

Structure, Dynamics and Vibrational Spectroscopy of Interfacial Alkali Nitrate Aqueous Solutions from $ab\ initio$ Molecular Dynamics

by Gang Huang

The interfacial structure and dynamics of solutions containing alkali nitrates have been studied by Density Functional Theory-based Molecular Dynamics (DFTMD) simulations. In particular we have presented a detailed analysis of the Hydrogen Bond (HB) structure at the interface and have calculated the interface vibrational spectra in order to provide a molecular interpretation of available experimental data. When calculating the Vibrational Sum-Frequency Generation (VSFG) spectrum, we used a new approximation method. This method ignores the NR part of the spectrum and finally obtains the interface's nonlinear polarizability to infrared incident light through the velocity Auto-Correlation Function (ACF). Based on the calculation of Maximally-Localized Wannier Functions (MLWFs), the parameterization of the spatial gradient of the components of dipole moment and the polarization is realized. Therefore, the calculation of the correlation between the dipole moment and the polarization is transformed into the calculation of the velocity ACF. The theoretical value of the imaginary part of the nonlinear polarizability obtained by this method is the same as the experimental Phase Sensitive Sum-Frequency Generation (PS-SFG) spectrum result.

As a first system we have analyzed the behaviour of a salty interface containing LiNO₃. Both the measured and calculated VSFG spectra shows a reduced intensity of the lower frequency portion region, when compared to the water/vapor interface. This reduction is attributed to the Hydrogen (H-) bonds established between the NO_3^- and the surrounding water molecules at the interface. This effects is only related to the presence of NO_3^- at the water surface and is not affected by the presence of alkali metal ions. I have shown that the use of simple models, such as small cluster is not suitable to reproduce the experimental spectra and cannot provide a microscopic interpretation of the spectra. Realistic models of the interface are required to address the perturbation of the ion at the water surface.

From the results of nonlinear susceptibilities, we conclude that water molecules at the interfaces of LiI, NaI, and KI solutions are participating in weaker H-bonds, compared with those at the water/vapor interface. The origin of the characteristics comes from a unique distribution of $\rm I^-$ ions and alkali metal cations, which form a double layer over the thickness on the order of 5–10 Å.

For the bulk and interfacial systems, we calculated the correlation functions of the HB population, and obtained the reaction rate constants with the HB formation and breakage. To analyze the HB dynamics at interfaces, we determined the instantaneous interface with Willard and Chandler's method based on spatial density, and proposed a new statistical scheme based on the interfacial HB population for the instantaneous interface. Combining this method with molecule sampling, we can get more realistic and accurate HB dynamics for the water/vapor interface. This method is also extended to general interface systems, such as the solvation shells of ions in aqueous solutions. The difference between the HB dynamics in the second solvation shell of the Li⁺ and that of nitrate-water Hydrogen (H-) bonds at interfaces is not visible from the values of the HB relaxation time. For the interface of alkaline iodine solutions, we also find that the cations does not alter the H-bonding network outside the first hydration shell of the cations themselves. Through solvation shell HB (SHB) dynamics, we obtained the HB dynamics for the H-Bonds between water molecules in the ion's second solvation shell. Similarly, we also calculated the reorientation dynamics $(C_2(t))$ of the water molecules in the first solvation shells of ions. We found that the reorientation dynamics for the water molecules in the first solvation shell is significantly affected by the ions. In contrast, the HB dynamics for the H-bonds between water molecules in the second solvation shell are not sensitive to the ions' nature.

Acknowledgements

I wish to express my sincere gratitude to my supervisor, Prof. Dr. Marilore Sulpizi for her excellent guidance during my doctoral studies, and her detailed and valuable advice for my writing. I thank her for bringing me into interesting and novel areas of density functional theory-based molecular dynamics simulation and interface physics and chemistry. By communicating with her, I was implicitly and positively affected. She is rigorous in academic studies, optimistic and active, and has a wide range of interests. All these have kept in my heart and gave me greater confidence in future scientific research work.

I wish to thank Prof. Dr. Thomas Kühner, Dr. Giovanni Settanni, PD Dr. Peter Virnau, Dr. Hans Behringer and Prof. Dr. Thomas Speck at the institute for physics for leading me to the fundamental statistical physics and excellent lectures and discussions. I am grateful to Prof. Dr. Friederike Schmid and Prof. Dr. Kurt Binder for their talks and guidance.

I thank Prof. Dr. Harvey Meyer at the institute for nuclear physics. I feel very honored to be a teaching assistant in his quantum field theory course. I have benefited a lot from his wonderful teaching methods, exercises, and questions and answers with students in his class.

I am grateful to Rémi Khatib for programing skills and helpful guidance on the programs for calculating the VSFG spectra. I also thank Shuanhu Qi, Jiajia Zhou and Fei Yu for useful discussions on the the calculation of response functions. I would like to thank Leila Salimi, Isidro Lorenzo Geada, Santosh Kumar Meena and Anusha Lalitha, at the Institute for Physics, for their encouragement and assistance.

I thank Astrid Chase and Daniela Reibel for their enthusiastic and patient help during my stay at the University of Mainz. Thanks to Andreas Nussbaumer for providing technical support so that I can carry out the calculation smoothly.

I thank Prof. Honggang Luo, Prof. Haiping Fang and Prof. Haijun Zhou for their great help in my study during my doctoral studies. The cooperation and discussions with Guosheng Shi, Junming Shao, Bo Zhou and Chunlei Wang have benefited me a lot. I express my sincere thanks to them.

I thank Prof. Yuliang Jin for his support, trust and teachings over the past year. I am grateful to my parents Yuhua Diao and Dehuai Huang, and my brother Jialiang, for their care, understanding and support for me for decades.

The financial supports of the China Scholarship Council and TRR146 are gratefully acknowledged.

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List of Abbreviations

ACF Auto-Correlation Function AIMD ab initio Molecular Dynamics

BLYP Beck-Lee-Yang-Parr

BOMD Born-Oppenheimer Molecular Dynamics
CPMD Car-Parrinello Molecular Dynamics

DFT Density Functional Theory

DFTMD Density Functional Theory-based Molecular Dynamics

GGA Generalised Gradient Approximation

H- Hydrogen

HB Hydrogen BondHK Hohenberg-Kohn

SHB Solvation shell Hydrogen Bond IHB Interface Hydrogen Bond

KS Kohn-Sham

LDA Local Density Approximation

MD Molecular Dynamics

MLWF Maximally-Localized Wannier Functions

MP2 Moller-Plesset Perturbation
PBE Perdew-Burke-Ernzherhof

PS Phase-Sensitive

RDF Radial Distribution Function

SCF Self Consistent Field

VDOS Vibrational Density Of States

VSFG Vibrational Sum-Frequency Generation

XC eXchange and Correlation

Physical Constants

Avogadro constant
Boltzmann constant
elementary charge
electron mass
molar gas constant
proton mass
Planck constant
reduced Planck constant
speed of light in vacuum

$$\begin{split} N_{\rm A} &= 6.022\ 140\ 76\ \times 10^{23}\ /{\rm mol}\ ({\rm exact})\\ k_{\rm B} &= 1.380\ 649\ \times 10^{23}\ {\rm J\,K^{-1}}\ ({\rm exact})\\ e &= 1.602\ 176\ 634\ \times 10^{-19}\ {\rm C}\ ({\rm exact})\\ m_{\rm e} &= 9.109\ 381\ 887\ \times 10^{-31}\ {\rm kg}\\ R &= 8.314\ 459\ {\rm 8\ J\ mol^{-1}\ K^{-1}},\\ m_{\rm p} &= 1.672\ 621\ 581\ \times 10^{-27}\ {\rm kg}\\ h &= 6.626\ 070\ 15\ \times 10^{-34}\ {\rm J\ s}\ ({\rm exact})\\ \hbar &= 1.054\ 571\ 817\ \times 10^{-34}\ {\rm J\ s}\\ c &= 299\ 792\ 458\ {\rm m\ s^{-1}}\ ({\rm exact}) \end{split}$$

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List of Symbols

 $I_{\rm SFG}$

 $I_{\rm vis}$

 f_i activity coefficient of species i $Z_{\rm X}$ atomic number of atom X \mathbf{R} atomic coordinates average lifetime of H-bonds: $\langle \tau_{\rm a} \rangle = \int_0^\infty t P_{\rm a}(t) dt$ chemical potential of component i μ_i decay rate of $C_2(t)$ κ_i coordination number for ion X $n_{\mathbf{X}}$ difference between the first peaks' positions of the RDFs g_{X-Q} and g_{X-H} δ_1 \mathbf{M} dipole moment $\mu^{i,l,\epsilon}$ dipole moment of the bond ϵ of the *i*-th water molecule in the lab frame dipole polarizability (tensor) ADdirection cosine matrix distance relectric dipole vector μ ϕ electric potential \mathbf{E} electric field strength $n(\mathbf{r})$ electron density $H_{\rm e}$ electronic many-body Hamiltonian $E_{\rm I}({\bf R})$ energy of the nuclei $V(\mathbf{r})$ external potential $V_{\rm ee}$ electron-electron repulsion energy T[n]electronic kinetic energy $E_{\text{disp}}^{(2)}$ $E_{\text{disp}}^{(3)}$ empirical two-body dispersion correction of energy empirical three-body nonadditivity dispersion correction of energy FFarady constant ΔF_{AB} free energy difference between configuration A and B F_A free energy of configuration A Fresnel coefficients $L_{\eta\kappa}$ frequency of the sum-frequency generation beam ω_{SFG} frequency of the incident visible beam ω_{vis} frequency of the incident infrared beam ω_{IR} h(t)HB population operator $h^{(d)}(t)$ HB population operator (it is 1 if the tagged pair are closer than r_{OO}^c) H(t)HB population operator (continuously) $C_{\mathrm{HB}}(t)$ (or c(t)) HB population auto-correlation function highest nuclear phonon frequency ω_{I} θ H-O-H angle in a water molecule hyperpolarizability; reciprocal temperature $h^s(t)$ interfacial HB population operator j(t)integrated flux departing the HB configuration space at time t

intensity of the sum-frequency generation beam

intensity of the incident visible beam

 $I_{\rm IR}$ intensity of the incident infrared beam

 $r_{\rm OO}$ interoxygen distance

 $\omega_{\rm e}$ the lowest electronic frequency

 $r_{\rm OO}^{\rm c}$ the maximum value of the interoxygen distance (in the definition of HB) the maximum value of the angle between the O–O axis and one of the

O–H bonds (in the definition of HB)

 m_i molality (bulk or surface) of the ion i

N number of water molecules R_{OH} O-H length in water molecule $C_2(t)$ orientational anisotropy decay

Z partition function α polarizability tensor

 $\alpha^{i,l,\epsilon}$ polarizability of the bond ϵ of the *i*-th water molecule in the lab frame

 $P_a(t)$ probability distribution of the first HB breaking in time t $C_z(t)$ projected velocity auto-correlation function along z-axis

 x^{1}, y^{1}, z^{1} position coordinates in the lab frame x^{m}, y^{m}, z^{m} position coordinates in the molecular frame x^{b}, y^{b}, z^{b} position coordinates in the bond frame

 $g_z(\nu)$ projected VDOS for selected atoms along z-axis

 g_{A-B} radial distribution function k rate constant of breaking a HB k' rate constant of reforming a HB

 $r_{\rm X}$ ratio of number of X to the number of $H_2{\rm O}$ among neighbors of a water molecule

 $\theta_{\rm SFG}$ reflected angle of SFG beam w.r.t. the normal direction in the medium

k(t) reactive flux

 $au_{
m R}$ relaxation time of HB population operator correlation function

 $P_2(x)$ the second Legendre polynomial $\chi^{(2),R}$ second-order resonant susceptibility $\chi^{(2),NR}$ second-order nonresonant susceptibility

 E_{KSDFT} self consistent KS energy $\Delta \nu'$ shift of vibrational frequency

 $h^{k,X}(t)$ solvation shell HB population operator

 m_i^* solvent molality stoichiometry of ion i

 Γ_i surface excess of component i

 $K_{\rm p,X}$ surface/bulk molar concentration ratio of ion X

 γ surface tension

 $S_{\rm HB}(t)$ (or s(t)) survival probability of HB

 t_t temporal resolutions for calculating HB dynamics a_i thermodynamic activity of a chemical species i

 K^{\ominus} thermodynamic equilibrium constant d thickness of layer of interface (model) τ time for switching allegiance of a HB $\hat{\mu}(t)$ unit vector of the transition dipole

 $g(\nu)$ VDOS for selected atoms z_i valence of species i velocity of an atom

C(t) velocity auto-correlation function

vibrational frequency; vibrational quantum number

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To my family.

Chapter 1

Introduction

Because of extremely high polarity, water is a unique solvent for salts. Interfaces of aqueous electrolyte solutions are ubiquitous in the biology, atmosphere, chemistry, man-made systems and industrial processes [1–14] Many phenomena in biology and chemistry, such as adsorption, bubble formation, occur at aqueous interfaces[15, 16]. Ions at the solution/vapor interface can undergo heterogeneous or interfacial reactions[17–21]. Compared to bulk atoms or molecules, interfacial atomic or molecular layers generally have very different optical properties. Therefore, study the structure and dynamics of aqueous interfaces (solution/vapor interfaces) is essential for obtaining the underlying mechanism of the experimental phenemena about them.

Experimentally, in order to determine the structure of an interface, one can use Vibrational Sum-Frequency Generation (VSFG) spectroscopy, which utilizes a secondorder nonlinear optical process and the resulting signal is very sensitive to surface ions and molecules of a submonolayer level [22–26]. This technique allows for detecting intramolecular vibrational modes, and molecular orientation by detecting polarization dependence of the VSFG signals [27]. Furthermore, the VSFG spectroscopy does not require ultrahigh-vacuum environment, because the interface selectivity is attributed to the symmetry reasons [28, 29]. The advantage is its wide applicability to almost every interface as long as light can reach them. Therefore, the VSFG spectroscopy can be used to probe many types of interfaces, namely, liquid-liquid and solid-liquid interfaces [6, 22, 30–35]. With the electric dipole approximation, the VSFG process is forbidden in any centrosymmetric bulk medium [36], such as isotropic liquids and glasses, but it is allowed at interfaces because of the broken inversion symmetry at the interfaces [37]. The VSFG spectroscopy can also be applied to metal and semiconductor surfaces [38, 39]. The VSFG spectra suggest that the interfacial H-bonding between water molecules is changed by the presence of salt, especially the anions [40]. Molecular-level properties of interfacial materials arising from interactions between water and minerals, such as swelling, wetting, hydrodynamics can also be studied by the VSFG spectroscopy[41].

However, the quantitative interpretation of the VSFG spectra is not straightforward, because the VSFG intensity is influenced by several factors, including ions' concentration, molecular orientation and distribution and local field correction[22]. Although there is some general consensus on the fact that anions propensity inversely correlates with the order of the Hofmeister series, namely $CO_3^{2-} > SO_4^{2-} > F^- > Cl^- > Br^- > NO_3^- > I^- > ClO_4^- > SCN^-[42-45]$, the driving force and the microscopic details of the solvation structure are still debated. In terms of the Gibbs adsorption equation, the traditional theory predicts that no atomic ions will exist at interfacial regions of solutions. However, Molecular Dynamics (MD) simulations have shown that more polarizable anions (e.g., larger halide anions) are present in the surface region[46, 47]. For example, Tian and coworkers [48] have predicted that some ions, such as I^- and Br^- , could accumulate at the interface. The surface tension has been

used to infer the composition of water/vapor interface, since the surface tension of the interface is generally altered by dissolved substance [49].

In recent years, MD simulations have been used for calculating properties, such as the depth profile of ion concentrations of interfaces[46, 47], and the VSFG spectra of electrolyte solution surfaces[25, 34, 50, 51], but the results depend heavily on the molecular model and interaction potentials used[52–55]. In this thesis, we use Density Functional Theory-based Molecular Dynamics (DFTMD) simulations to compute the interfacial VSFG spectra of eletrolyte solutions and to provide their molecular interpretation. The advantage of DFTMD is that it does not require a priori parameterization and it is capable to include polarization effects[56], also including electronic polarization. DFTMD at the gradient corrected level, and also including dispersion corrections [57–61] has been shown to provide an accurate description of the vibrational properties at interface[62]. They are used in DFTMD simulations.

Recently considerable attention has been given to the nitrate ions in aqueous phase for their ubiquitous and diverse role in atmospheric aerosols, polluted water, and the remote troposphere [63, 64]. In order to clarify the variations of the structural and dynamical properties of water containing ions with high surface propensity, MD simulations are a valuable tool, which can provide detailed information on the structure and dynamics of water in a simple and well-known system like alkali metal nitrate and alkali metal halide solutions [65]. We are concentrated on the liquid/water interface of alkali metal nitrate and alkali metal halide solutions.

The hydrogen (H-) bonding network [66-72] as well as electrostatic force, and van der Waals force are the main factors that determine the structure of interfaces. Salts change the H-bonding structure of water in the interfacial region[15, 40, 73]. The difference between anions' distribution may have significant influence on the H-bonding network of interfacial water [22]. Hydrogen bond (HB) dynamics is studied for the water/vapor and solution/vapor interfaces to obtain the structural characteristics of the solution/vapor interfaces. Due to molecular motions, the identity of molecules that on the interface also changes over time [74], and the fixed interface will no longer apply. To represent the interface at molecular level, the technique for calculating instantaneous liquid interface are used to define a liquid interface from atomic coordinates. The instantaneous interfaces are calculated for the aqueous electrolyte solutions, and are used in the analyse of interfacial HB dynamics, reorientation relaxation rate of water molecules, etc. Based on the instantaneous interface layer, we obtained a series of properties about the aqueous interfaces that are consistent with the experimental results, such as the thickness of the water/vapor interface, the relaxation time of the hydrogen bond at the interface, and the distribution of free OH bonds at the interface that have an important contribution to the VSFG spectrum. These results suggest that considering instantaneous interfaces is essential for understanding the experimental results about aqueous interfaces.

The thesis is organized as follows. In chapter 2, we present the methods of *ab initio* Molecular Dynamics and one of its implementations—the DFT and the method to calculate the VSFG spectra, and showed the available experimental data for the VSFG spectra of salty interfaces. In chapter 3, we present the results of the vibrational properties of water clusters including alkali metal nitrates, which are calculated for interpreting the vibrational characteristics of water molecules in a special water/vapor interface—the water clusters. The theoretical results of VSFG spectra of solution/vapor interfaces of alkali metal nitrate solutions are included in chapter 4. Chapter 5 focuses on the HB dynamics of the water/vapor interface, and introduce the method for studying the HB dynamics of the instantaneous interface layer of the water/vapor interface. In addition to the alkali nitrate solutions, in chapter 6, we

also simulated and analyzed the alkali iodide solutions and its interface, which have similar properties in many aspects. The conclusions are summarized in chapter 7.

Chapter 2

Methods

In this chapter, we describe the methods to study the structure and dynamics of water and aqueous electrolyte solutions. We use DFTMD simulations to calculate the theoretical VSFG spectra. Our motivations for adopting DFTMD simulations are manifold. In general, by using DFTMD, (1) Structure and reactivity can be treated in a consistent way; (2) Efficient treatments of basis sets and long range interactions in the DFT do extend the simulation capabilities to thousands of atoms, i.e. it allows realistic models for interfaces; (3) They provide information on the structural organization of the solvent in the interfaces. In particular, the interface of the solution contains a large number of water molecules and a HB network composed of H-bonds between them. As Stillinger said, since H-bonding is the most important interaction in liquid water and since these interactions are cooperative, it is insufficient for the purposes of computer simulation to use the potential energy function for dimers alone [75].

The organization of this chapter is the following. Paragraph 2.1 to 2.3 give an introduction to the basic ideas of AIMD and DFT as well as the BOMD we adopted. Paragraph 2.4 and 2.5 introduce the analysis methods used in this paper, including the method of calculating vibrational properties from velocity autocorrelation and the method of calculating SFG spectrum from velocity correlation function. Paragraph 2.6 includes the available experimental data for the VSFG spectra of salty interfaces.

2.1 Modeling interfaces with *ab initio* molecular dynamics

Modern theoretical methodology, aided by the advent of high speed computing, has advanced to a level where the microscopic details of dynamical processes in condensed phases can be treated on a relatively routine basis. One common theoretical approaches for obtaining these microscopic details of the system is the MD method. In the MD simulations, the classical Newtonian equations of motion for a system are solved numerically starting from a prespecified initial state, and subject to a set of boundary conditions appropriate to the problem. The MD methodology allows both equilibrium thermodynamic and dynamical properties of a system at finite temperature to be computed, while simultaneously providing a view of the microscopic motion of individual atoms in the system [76].

Despite the success of classical molecular dynamics simulations, they have some limitations. First, charges are treated as static parameters, therefore electronic polarization effects are not included. The so-called polarizable models[77–79], in which charges and induced dipoles are allowed to fluctuate in a changing environment,

have been proposed to overcome this problem. While they have considerable success, they also have serious limitations, including a lack of transferability and standardization[80]. Second, force fields assume a pre-specified connectivity among the atoms, therefore, they suffer from an inability to describe chemical bond-forming and -breaking. This problem can be treated using techniques such as the empirical valence bond method [81] or other semi-empirical approaches. Unfortunately, these methods are also not transferable and, therefore, need to be reparametrized for each type of reaction and may end up biasing the reaction path in undesirable ways.

To overcome these limitations of force field based approaches, *ab initio* molecular dynamics (AIMD) simulation techniques [82–88] can be used. The AIMD combines finite temperature dynamics with forces obtained from electronic structure calculations performed 'on the fly' as the MD simulation proceeds [87]. Because the electronic structure is treated explicitly in the AIMD calculations, many-body forces, electronic polarization and bond-forming and -breaking events are described with the accuracy of the electronic structure representation. Moreover, the AIMD method can be easily extended to incorporate nuclear quantum effects via the Feynman path integral approach [89, 90], leading to the *ab initio* path integral technique [91–93].

The AIMD method have been used to study a wide variety of chemically interesting problems in areas such as chemical reactivity, H-bonds for the interfacial structure, pKa, and vibrational spectroscopy. AIMD applications include calculations of the structure and dynamics of water and other H-bonded liquids, proton transport in aqueous and condensed phase environments, structure, proton order/disorder and dynamical properties of ice, structure of liquid silicates and glasses, mechanisms of polymer knotting, Ziegler-Natta industrial catalysis and other surface catalytic processes. More recently, the AIMD methods have started to impact the biological sciences and have been applied in calculations of nuclear magnetic resonance chemical shifts in drug-enzyme complexes, structure of nucleic acids, exploration of the design of possible biomimetics and structure, dynamics and binding mechanisms in myoglobin. In many of these applications, new physical phenomena have been revealed, which could not have been uncovered using empirical models, often leading to new interpretations of experimental data and even suggesting new experiments to perform [80].

To study the heterogeneous environment at water interface, AIMD is particularly suitable for the following reasons: (1) AIMD is only relying on the atomic coordinates of the model system \mathbf{R} , and not on any adjustable parameter, i.e., the interatomic forces $\mathbf{F}_{\rm I} = -\nabla_{\mathbf{R}_{\rm I}} V(\mathbf{R})$, where $V(\mathbf{R})$ is the potential energy[94], are determined using the first principle electronic structure methods on the fly; (2) New phenomena that are not foreseen before starting the simulation can simply happen if necessary. Therefore, AIMD is also a good predictive tool. However, as a drawback, the AIMD simulations are expensive and can nowadays be performed on size-limited system of 100–1000 atoms for up to a few hundred ps.

2.2 Density functional theory

In most currently performed AIMD simulations, the dynamics is performed within the so-called Born-Oppenheimer (BO) approximation. Since the mass of an electron is much smaller than that of any nuclei, there is a strong separation of timescales between the electronic and nuclear motion. We assume that the kinetic energy of nuclei is zero and their potential energy is constant in each moment of the dynamics. Therefore, the electrons can be treated independently at constant nuclear coordinates **R**. Applying BO approximation, the potential energy $V(\mathbf{R})$ is written as

$$V(\mathbf{R}) = \langle \Psi_0 | H_e | \Psi_0 \rangle + E_I(\mathbf{R}), \tag{2.1}$$

where $|\Psi_0\rangle$ is the ground state, $E_I(\mathbf{R})$ is the energy of the nuclei, and H_e is the electronic many-body Hamiltonian, which depends on the electronic coordinates but parametrically on the nuclear degrees of freedom,

After the BO approximation, a formidable task is to solve the electronic, non-relativistic, time independent many-body Schödinger equation

$$H_{\rm e}|\Psi_0\rangle = \epsilon_0(\mathbf{R})|\Psi_0\rangle.$$
 (2.2)

This is a high-dimensional eigenvalue problem, and it is still time consuming. One great solution to the electronic structure problem (to solve the electronic, non-relativistic, time independent many-body Schödinger equation) is the DFT[95, 96].

DFT can be used to map the problem of a interacting electron gas onto that of a single particle in an effective non-local potential [83]. It provides a favorable compromise between computational cost and accuracy.

Hohenberg and Kohn (HK) proved that the total energy of a many-electron system is a unique functional of the electron density $n(\mathbf{r})$. The first HK theorem proves that there exists a one-to-one correspondence between the ground state electronic density $n_0(\mathbf{r})$ and an external potential $v(\mathbf{r})$, i.e., the electron density n determines all properties of a non-degenerate ground state of an atom or molecule (for a degenerate ground state the density n determines the energy). (If we want to solve variationally for the ground state energy of a system with $H = T + V_{ee} + \sum_{i=1}^{N} v(i)$) the HK theorem says that there exists a valid functional Q[n] that delivers the sum of the electronic kinetic energy T[n] and electron-electron repulsion energy $V_{ee}[n]$ of each trial electron density n[97].

The electronic density $n(\mathbf{r})$ which depends on just 3 electronic degrees of freedom, become the central quantity in DFT in place of the complex $3N_{\rm e}$ -dimensional many-body wave-function.

The second HK theorem states that the total energy in the electronic density space satisfies

$$E^{\text{DFT}}[n_0] = \psi_0 H_e \psi_0 \le \psi' H_e \psi' = E^{\text{DFT}}[n'],$$
 (2.3)

for which the equality holds iff $n_0 = n'$. (variational principle) One try to choose different n to optimize $E^{\text{DFT}}[n]$, the quantum expectation value of $H_{\text{e}}[n]$, thus to determine $E^{\text{DFT}}[n_0]$, i.e.

$$E^{\text{DFT}}[n_0] = \min_{\psi} \psi H_e \psi = \min_{n} \psi[n] H_e[n] \psi[n] = \min_{n} E^{\text{DFT}}[n],$$
 (2.4)

Eq. (2.2) can be solved by iteratively diagonalizing $H_e[n]$ within a self consistent field (SCF) procedure.

Assuming atomic units and considering the physical relevant Coulomb interaction, the total Hamiltonian is

$$H_{e} = \frac{1}{2} \sum_{i=1}^{N_{e}} \nabla_{i}^{2} + \sum_{i < j}^{N_{e}} \frac{1}{|\mathbf{r}_{i} - \mathbf{r}_{j}|} + \sum_{I,i}^{N,N_{e}} \frac{Z_{I}}{|\mathbf{R}_{I} - \mathbf{r}_{i}|}$$
$$= \hat{T} + \hat{U} + \hat{V}, \tag{2.5}$$

where Z_I is the atomic number of atom I, \hat{T} is the operator of kinetic energy of

electrons, \hat{U} is the electron-electron interaction and $\hat{V} = \sum_{i} v(\mathbf{r}_{i})$ is the electron-nuclei operator.

In DFT, we obtain the ground state energy of a many-electron system as minimum of an energy functional

$$E^{\text{DFT}}[n(\mathbf{r})] = T[n(\mathbf{r})] + U[n(\mathbf{r})] + V[n(\mathbf{r})]. \tag{2.6}$$

The next problem is how to provide an explicit form for the three terms appearing in Eq. (2.6). The solution is the Kohn-Sham approach to DFT. A reference system with the same electron density as the density for the full interacting system and without electron electron repulsion is introduced. For such reference system, the kinetic energy functional is

$$T_{\mathrm{s}}[n(\mathbf{r})] = -\frac{1}{2} \sum_{i=1}^{N_{\mathrm{e}}} \int d\mathbf{r} \psi_i^*(\mathbf{r}) \, \nabla^2 \, \psi_i(\mathbf{r}) = T_{\mathrm{s}}[\{\psi_i[n(\mathbf{r})]\}], \tag{2.7}$$

and the electronic density can be written as

$$n(\mathbf{r}) = \sum_{i=1}^{N_{\text{occ}}} f_i \psi_i(\mathbf{r}) \psi_i^*(\mathbf{r}), \qquad (2.8)$$

in which N_{occ} is the number of occupied orbitals and f_i the occupation number of the ith state, so that

$$\sum_{i=1}^{N_{\text{occ}}} f_i = N_{\text{e}},\tag{2.9}$$

and $\psi_i(\mathbf{r})$ is the wave-function of the *i*-th state.

The total energy functional of the Kohn-Sham (KS) system is

$$E^{KS}[n(\mathbf{r})] = E^{KS}[\{\psi_i[n(\mathbf{r})]\}]$$

$$= T_s[\{\psi_i[n(\mathbf{r})]\}] + U_H[n(\mathbf{r})] + V[n(\mathbf{r})] + E_{XC}[n(\mathbf{r})]$$

$$= -\frac{1}{2} \sum_{i=1}^{N} f_i \int d\mathbf{r} \psi_i^*(\mathbf{r}) \nabla^2 \psi_i(\mathbf{r}) + \frac{1}{2} \int d\mathbf{r} \int d\mathbf{r}' \frac{n(\mathbf{r})n(\mathbf{r}')}{\mathbf{r} - \mathbf{r}'}$$

$$+ \int d\mathbf{r} v_{\text{ext}}(\mathbf{r})n(\mathbf{r}) + E_{XC}[n(\mathbf{r})],$$
(2.11)

where $E_{\rm XC}[n(\mathbf{r})] \equiv (T[n(\mathbf{r})] - T_{\rm s}[\{\psi_i[n(\mathbf{r})]\}]) + (U[n(\mathbf{r})] - U_H[n(\mathbf{r})])$ is the exchange and correlation (XC) energy functional and $v_{\rm ext} = \delta V[n(\mathbf{r})]/\delta n(\mathbf{r})$ is the external potential. The energy functional $E_{\rm XC}[n(\mathbf{r})]$ is the only unknown part of the total energy functional. This definition of XC energy functional shows that a significant of part $E_{\rm XC}$ is due to correlation effects of the kinetic energy, that is expressed explicitly only with the reduced 2-particle density matrix.

Directly minimizing Eq. (2.10) is not straightforward because $T_s[\{\psi_i[n(\mathbf{r})]\}]$ is an explicit orbital functional. However, it is more appropriate to make $E_{KS}[\{\psi_i[n(\mathbf{r})]\}]$ stationary by the following Euler-Lagrange equation. It is possible to use the variational principle to derive the corresponding Euler-Lagrange equation of the non-interacting system within the potential v_{ext} . The KS scheme permits to map the full interacting many-body problem, with the electron-electron interaction \hat{U} onto an equivalent fictitious single-body problem, with an effective potential operator $\hat{V}_{\text{KS}} = \hat{U}_s + \hat{V}_{\text{H}} + \hat{V}_{\text{XC}}$.

If we figure out a way to approximate the $V_{\rm XC}$ accurately, we will have a much

less demanding set of equations to solve than those of the true system[98]. Using the variational principle is possible to derive the corresponding Euler-Lagrange equation of the reference system (non-interaction system) within the potential v. To determine the set of wave-functions ψ_i which minimize the KS energy functional, we can iteratively solve the equation

$$[T_{s}[n(\mathbf{r})] + V_{H}[n(\mathbf{r})] + V[n(\mathbf{r})] + E_{XC}[n(\mathbf{r})]]\psi_{i}(\mathbf{r}) = \varepsilon_{i}\psi_{i}(\mathbf{r}), \qquad (2.12)$$

where ε_i is the eigenvalue of each equation, $V_{\rm H}$ is the Hartree potential of the electrons:

$$V_{\rm H} = e^2 \int \frac{n(\mathbf{r})n(\mathbf{r'})}{|\mathbf{r} - \mathbf{r'}|} d\mathbf{r'}, \qquad (2.13)$$

and the exchange and correlation potential

$$V_{\rm XC} = \frac{\delta E_{\rm XC}[n(\mathbf{r})]}{\delta n(\mathbf{r})}.$$
 (2.14)

Given the explicit form of the exchange and correlation functional $E_{\rm XC}$, the exchange and correlation potential $V_{\rm XC}$ can be determined, and thus electron density $n(\mathbf{r})$. Thus the KS equation must be solved self-consistently.

The simplest density functional approximation is the Local Density Approximation (LDA). In LDA, the exchange and correlation energy of an electronic system is

$$E_{\rm XC}[n(\mathbf{r})] = \int \varepsilon_{\rm XC}(\mathbf{r})n(\mathbf{r})d\mathbf{r}, \qquad (2.15)$$

and

$$\frac{\delta E_{\rm XC}[n(\mathbf{r})]}{\delta n(\mathbf{r})} = \frac{\partial n(\mathbf{r})\varepsilon_{\rm XC}(\mathbf{r})}{\partial n(\mathbf{r})},\tag{2.16}$$

where

$$\varepsilon_{\rm XC}(\mathbf{r}) = \varepsilon_{\rm XC}^{\rm hom}[n(\mathbf{r})].$$
 (2.17)

The LDA is accurate for systems with slowly varying charge densities. It has a tendency to favor more homogeneous systems and over-binds solids and molecules. The dielectric and piezoelectric constant calculated from the LDA are approximately 10% over estimated. The limitations of the LDA suggest that care must be taken into its applications. For example, the independent particle picture breaks down in strongly correlated systems, where the LDA is very inaccurate. Furthermore, the LDA does not take into account variation of the electronic density and van der Waals interactions[98], so it does not give a very accurate description of H-bonding. H-bonding is essential for a correct description of water and interfaces with water, therefore, a functional beyond the LDA is needed in the description of hydrogen-bonded systems, including water.

For any density that varies sufficiently slowly, an expansion of a functional f in gradients should have increasing accuracy:

$$f[n] = \int d^3r [an(\mathbf{r}) + bn(\mathbf{r})|\nabla n(\mathbf{r})|^2 + \cdots]. \tag{2.18}$$

But the gradient expansion of the exchange-correlation energy does not always improve results, sometimes it leads to divergences. Therefore, a more general approach which is called Generalised Gradient Approximation (GGA) is considered. An approach to improve the LDA is to include gradient corrections, in which $E_{\rm XC}$ is a

functional of density and its gradient:

$$E_{\rm XC}^{\rm GGA}[n(\mathbf{r})] = \int \varepsilon_{\rm XC}(n(\mathbf{r})) n(F_{\rm XC}[n(\mathbf{r}), |\nabla n(\mathbf{r})|]) d\mathbf{r}, \qquad (2.19)$$

where $F_{\text{XC}}[n(\mathbf{r})]$ is a correction chosen to satisfy conditions for E_{XC} . The XC energy depends locally on the gradient of the density ∇n as well as the density n. There are several forms of the GGAs.

The GGA is usually the best compromise between speed and accuracy in large systems. For solids, the most commonly used functional with the GGA is the one proposed by Perdew, Burke and Ernzherhof (PBE)[99]. Another popular GGA functional is the BLYP functional. The electron densities of electric dipole and quadruple moments are not uniform in space. Physically, GGAs include information on the spatial variation in the electron densities, and thus they can create functionals with better flexibility to describe dipole and quadruple moments. Therefore, GGAs generally describe the dipole and quadruple moments of the monomer quite well. However, they somewhat overestimate polarizabilities, the predicted dipolar polarizability (also called dipole polarizability) being typically 10% too large.

Dispersion is a general term referring to weak, long-range correlations in electronic structure. It includes van der Waals interactions, which originates from the coupling of the electric field generated by fluctuations in the electronic density at position $\bf r$ in the system with the density at another point $\bf r'[100,\ 101]$. These interactions are not well modelled by any mean-field level of theory, the *ab initio* wave-function theory such as second-order Moller-Plesset perturbation theory (MP2), or standard DFT functionals, eg. LDA, GGAs, etc. The mean-field theory does not include the electron correlation effect, MP2 theory usually overestimate the binding energies and underestimates intermolecular equilibrium distances, and all the gradient corrected DFT are unable to describe dispersion interactions, because they can not describe the long-range electron correlation. In the last two decades, a series of empirical corrections have been proposed which can improve the structural properties without more computational cost[57–60, 102–104]. Among such empirical approaches we choose the DFT-D3 correction which can be used in our application to interfaces with water [60, 105].

In the DFT-D3 correction, also used in this thesis, the input parameters are cutoff radii and dispersion coefficients, and they can be calculated by KS-DFT methods using extended atomic orbital basis sets. The use of structure dependent dispersion coefficients, i.e., functional coordination number, to interpolate between dispersion coefficients of atoms in different chemical environments, increases the accuracy. Moreover, no atom connectivity information is required and all the properties are calculated only from Cartesian coordinates and atomic numbers[60].

If the three-body nonadditivity terms are considered, as well as the pairwise terms, the total energy is given by

$$E_{\text{DFT}} = E_{\text{KS-DFT}} + E_{\text{disp}}^{(2)} + E_{\text{disp}}^{(3)},$$
 (2.20)

where $E_{\text{KS-DFT}}$ is the usual self-consistent KS energy obtained from the chosen density functional, $E_{\text{disp}}^{(2)}$ is the empirical two-body dispersion correction term and $E_{\text{disp}}^{(3)}$ is the three-body nonadditivity term. $E_{\text{disp}}^{(2)}$ is given by

$$E_{\text{disp}}^{(2)} = \sum_{AB} \sum_{n=6.8.10,\dots} s_n \frac{C_n^{AB}}{R_{AB}^n} f_{\text{dmp},n}(r_{AB}), \qquad (2.21)$$

where \sum_{AB} denotes the sum over all atom pairs in the system[100], C_n^{AB} is the averaged n-order dispersion coefficients ($n = 6,8,10,\cdots$) for atom pair AB, r_{AB} is the internuclear distance of atom pair AB, and $f_{dmp,n}(r_{AB})$ is the damping function used to avoid near-singularities for small distance R between nuclei. The damping function in Eq. (2.21) is given by

$$f_{\text{dmp},n}(r_{\text{AB}}) = \frac{1}{1 + 6(r_{\text{AB}}/(s_{r,n}R_0^{\text{AB}}))^{-\alpha_n}},$$
 (2.22)

where $s_{r,n}$ is the order-dependent scaling factor of the cutoff radii R_0^{AB} , which is the most important parameter that has to be adjusted for each density functional and α_n is a parameter which can be adjusted manually such that the dispersion correction is smaller than 1% of $\max(|E_{\text{disp}}|)$ for typical covalent bond distances. In Ref.[60] $s_{r,6}$ is optimized by a least-squares fitting procedure and $s_{r,8}$ is fixed to 1 for all density functionals.

The leading non-additive dispersion term for three atoms A, B and C is

$$E_{\text{disp}}^{(3)} = \frac{C_9^{\text{ABC}}(3\cos\theta_a\cos\theta_b\cos\theta_c + 1)}{(r_{\text{AB}}r_{\text{BC}}r_{\text{CA}})^2},$$
(2.23)

where θ_a , θ_b and θ_c are the internal angles of the ΔABC , and C_9^{ABC} is the triple-dipole constant defined by

$$C^{\text{ABC}} = \frac{3}{\pi} \int_0^\infty \alpha^{\text{A}}(i\omega) \alpha^{\text{B}}(i\omega) \alpha^{\text{C}}(i\omega), \qquad (2.24)$$

which can be approximated by a geometric mean

$$C^{\text{ABC}} \approx -\sqrt{C_6^{\text{AB}} C_6^{\text{AC}} C_6^{\text{BC}}},\tag{2.25}$$

since the total three-body contribution is typically 5–10% of $E_{\rm disp} = E_{\rm disp}^{(2)} + E_{\rm disp}^{(3)}$.

2.3 Born-Oppenheimer molecular dynamics

In computational material science, the most popular AIMD simulation methods are the Born-Oppenheimer MD (BOMD) and Car-Parrinello MD (CPMD) methods. In the BOMD, the potential energy $E[\{\psi_i\}; \mathbf{R}]$ is minimized at each MD step with respect to $\{\psi_i(\mathbf{r})\}$ under the orthonormality condition

$$\langle \psi_i(\mathbf{r}) | \psi_j(\mathbf{r}) \rangle = \delta_{ij}.$$
 (2.26)

Thus the Lagrangian density is

$$L_{\text{BO}}(\{\psi_i\}; \mathbf{R}_I) = \frac{1}{2} \sum_{I=1}^{N} M_{\text{I}} \dot{\mathbf{R}}_I^2 - \min_{\{\psi_i\}} E[\{\psi_i\}; \mathbf{R}_I] + \sum_{i,j} \Lambda_{ij} (\langle \psi_i | \psi_j \rangle - \delta_{ij}), \qquad (2.27)$$

in which Λ is the Hermitian Lagrangian multiplier matrix. By the Euler-Lagrange equations one obtains the equations of motion

$$M_{I}\ddot{\mathbf{R}}_{I} = -\nabla_{R_{I}}\left[\min_{\{\psi_{i}\}} E[\{\psi_{i}\}; \mathbf{R}_{I}]|_{\langle\psi_{i}|\psi_{j}\rangle = \delta_{ij}}\right]$$

$$= -\frac{\partial E}{\partial \mathbf{R}_{I}} + \sum_{i,j} \Lambda_{ij} \frac{\partial}{\partial \mathbf{R}_{I}} \langle\psi_{i}|\psi_{j}\rangle$$

$$-2\sum_{i} \frac{\partial \langle\psi_{i}|}{\partial \mathbf{R}_{I}} \left(\frac{\delta E}{\delta \langle\psi_{i}|} - \sum_{j} \Lambda_{ij}|\psi_{j}\rangle\right)$$
(2.28)

The term $-\frac{\partial E}{\partial \mathbf{R}_{\mathbf{I}}}$ is the Hellmann-Feynman force, and the term $\sum_{i,j} \Lambda_{ij} \frac{\partial}{\partial \mathbf{R}_{\mathbf{I}}} \langle \psi_i | \psi_j \rangle$, i.e., the wave-function force $F_{\mathrm{WF}}[106]$, is a constraint force due to the orthonormality constraint. The last term comes from the fact that there is always an implicit dependence on the atomic positions through the expansion coefficient $c_{ij}(\mathbf{r})$ that is defined by

$$\psi_i(\mathbf{r}) = \sum_j c_{ij}(\mathbf{r})\phi_i(\mathbf{r}), \qquad (2.29)$$

where the KS orbitals are assumed to be real.

The CPMD is an alternative method to the BOMD, which includes the electrons in a single state [107]. In the CPMD, a coupled electron-ion dynamics is performed. The CP Lagrangian is

$$L_{\text{CP}}(\{\psi_i\}; \mathbf{R}, \dot{\mathbf{R}}) = \frac{\mu}{2} \sum_{i} \langle \dot{\psi}_i | \dot{\psi}_i \rangle + \frac{1}{2} \sum_{I=1}^{N} M_{\text{I}} \dot{\mathbf{R}}_{\text{I}}^2 - E[\{\psi_i\}; \mathbf{R}] + \sum_{i,j} \Lambda_{ij} (\langle \psi_i | \psi_j \rangle - \delta_{ij}), \qquad (2.30)$$

where the electronic degrees of freedom carries a fictitious mass parameter μ , and are characterized by orbital velocities $\{\dot{\psi}_i\}$. Applying the Euler-Lagrangian equations leads the equations of motion:

$$M_{\rm I}\ddot{\mathbf{R}}_{\rm I} = -\nabla_{R_{\rm I}} \left[E[\{\psi_i\}; \mathbf{R}] |_{\langle \psi_i | \psi_j \rangle = \delta_{ij}} \right]$$

$$= -\frac{\partial E}{\partial \mathbf{R}_{\rm I}} + \sum_{i,j} \Lambda_{ij} \frac{\partial}{\partial \mathbf{R}_{\rm I}} \langle \psi_i | \psi_j, \rangle \qquad (2.31)$$

$$\mu \ddot{\psi}_i(\mathbf{r}, t) = -\frac{\partial \langle \psi_i |}{\partial \mathbf{R}_{\rm I}} \left(\frac{\delta E}{\delta \langle \psi_i |} + \sum_j \Lambda_{ij} | \psi_j \rangle \right)$$

$$= -\hat{H}_{\rm e} \langle \psi_i | + \sum_j \Lambda_{ij} | \psi_j \rangle, \qquad (2.32)$$

where $-\delta E/\delta \psi_i$ are the electronic forces to propagate the electronic degrees of freedom in time within a fictitious Newtonian dynamics. No SCF cycle is required to quench the electrons to the Born-Oppenheimer surface and to force them to evolve adiabatically with respect to the nuclei.

To ensure the adiabatic energy-scale separation of the nuclear and the electronic degrees of freedom, i.e. to prevent energy transfer between them, the highest nuclear phonon frequency $\omega_{\rm I}$ must be much smaller than the lowest electronic phonon frequency $\omega_{\rm e}$. The condition $\omega_{\rm I} \ll \omega_{\rm e}$ is ensured by a proper choice of the fictitious mass μ . The fictitious mass determines the computational accuracy.

Which method is to favor depends largely on the definition of accuracy, as well as on the particular application. In the applications in this thesis, we use the BOMD method, as implemented in CP2K package [108, 109]. In the BOMD (1) the nuclear positions are propagated in time followed Newton's equations of motion with the electronic ground state energy as the potential energy surface; (2) the time evolution of the atomic coordinates is performed with the velocity Verlet algorithm [110].

Since there is no explicit electron dynamics, the maximum integration time step is simply given by the one intrinsic to nuclear motion, i.e., $\tau_e^{\rm BO} \approx \tau_{\rm n}$. In BOMD, the time step can be even larger if the nuclear dynamics becomes fairly slow[111]. But in order to resolve vibrations in molecular systems, the time step is decreased to less than 1 fs in this thesis[87].

2.4 Vibrational density of states

In order to obtain information on the vibrational properties of a molecular systems, it is possible to use the velocities Auto-Correlation Functions (ACFs), as calculated from the AIMD trajectories. The obtained Vibrational Density of States (VDOS) can provide information on the local environments on the OH-stretching mode of water molecules. For a system comprised of N atoms, the velocity ACF C(t) for a molecule is [54, 112, 113]

$$C(t) = \frac{\langle \sum_{i=1}^{N} \mathbf{v}_i(t) \cdot \mathbf{v}_i(0) \rangle}{\langle \sum_{i=1}^{N} \mathbf{v}_i(0) \cdot \mathbf{v}_i(0) \rangle},$$
(2.33)

where $\langle \dots \rangle$ denotes the average over starting times, t is the time interval, and \mathbf{v}_i denotes the velocity of the i-th atom. The VDOS $g(\nu)$ for selected atoms, which is a function of the vibrational frequency ν of atoms, is expressed in terms of the Fourier Transform (FT) of the velocity ACF of the atoms[114]. In equilibrium, C(-t) = C(t), and $g(\nu)$ is a real function, i.e.,

$$g(\nu) = \sqrt{\frac{2}{\pi}} \int_0^\infty dt \cos(2\pi\nu t) C(t). \tag{2.34}$$

Single components of the velocities ACFs can also be calculated. For example in the case of an interface it may be of interest to look at z-component (where z is the direction perpendicular to the Gibbs interface), we can calculate the Fourier transform $g_z(\nu)$ of the ACF of the vertical component of atomic velocity. $g_z(\nu)$ can be calculated by Eq. (2.34) if we replace C(t) by

$$C_z(t) = \frac{\langle \sum_{i=1}^N v_{i,z}(t)v_{i,z}(0) \rangle}{\langle \sum_{i=1}^N \mathbf{v}_i(0) \cdot \mathbf{v}_i(0) \rangle},$$
(2.35)

where $v_{i,z}$ is the vertical component of the velocity of the *i*-th atom. The VDOS $g_z(\nu)$ for water molecules is used to extract the O-H stretch along the vertical direction to the interface.

2.5 Calculation of VSFG spectroscopy for interfaces

As we introduced in Chapter 1, the VSFG spectroscopy is a powerful tool for extracting structural and dynamical information on surfaces and interfaces. It can be applied to any interface as long as light can reach it, also including liquid/metal interfaces[115–119] and buried interfaces[120, 121]; It can probe liquid interface with

molecular sensitivity[122–124]. Therefore, the VSFG spectroscopy can yield structural information about solution/vapor interfaces which could not be obtained with other techniques.

2.5.1 Nonlinear susceptibility of solution/vapor interfaces

VSFG The VSFG is a coherent nonlinear optical process in which incident laser beams with frequency and wave vector (ω_1, \mathbf{k}_1) and (ω_2, \mathbf{k}_2) produce a signal with frequency and wave vector $(\omega_1 + \omega_2, \mathbf{k}_1 + \mathbf{k}_2)$ (Fig. 2.1). In this process, the fields arise from the polarization induced over a macroscopically large volume superposition coherently. The VSFG signal intensity is proportional to the square of the resonant and non-resonant terms:

$$I_{SFG}(\omega) \propto |\chi^{(2)}(\omega)|^2$$

= $|\chi^{(2),R}(\omega) + \chi^{(2),NR}(\omega)|^2$, (2.36)

where $\chi^{(2)}$ is a third-rank tensor called second-order nonlinear susceptibility, or hyperpolarizability for a molecule[125], and it is responsible to the VSFG process. $\chi^{(2),R}$ is the vibrationally resonant term, while $\chi^{(2),NR}$ is the nonresonant term. Because

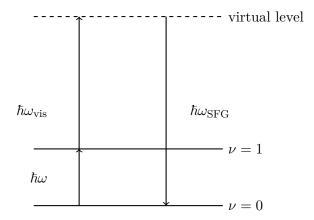


FIGURE 2.1: The schematic of the VSFG process which involves IR and Raman transitions. The $\nu=0,\,\nu=1$ levels denote the ground and the first excited state of the oscillator[126], respectively. The dashed line denotes a virtual electronic state in the Raman transition.

experiments usually employed visible and the VSFG frequencies are far from resonant conditions, $\chi^{(2),\rm NR}$ can be considered totally off-resonant and therefore insensitive to the laser beams' frequencies involved. Therefore, we can neglect the frequency dependence of the non-resonant term. The molecular information is contained in the resonant signal. The resonant susceptibility $\chi^{(2),\rm R}(\omega)$ is given by

$$\chi_{\eta\xi\kappa}^{(2),R}(\omega) = \frac{-i}{\hbar} \int_0^\infty dt e^{i\omega t} \text{Tr}[\rho, \mu_\kappa] \alpha_{\eta\xi}(t), \qquad (2.37)$$

where the index η , ξ and κ are one of x,y and z labels of the laboratory coordinate frame. In Eq. (2.37) $\rho = e^{-\beta H/Z}$ for a system with Hamiltonian H and partition function Z at reciprocal temperature $\beta = 1/k_BT$; μ_{κ} is the κ -th component of the system electric dipole and $\alpha_{\eta\xi}$ is the $\eta\xi$ -th component of the polarizability tensor[127]. Besides the vibrational resonance, $\chi^{(2),R}$, which reflects the vibrational and orientational characteristics of the surface species, the VSFG signal also includes the contribution

from the non-resonant signal background $\chi^{(2),\mathrm{NR}}$, due to static hyperpolarizability of the interface itself[36]. For example, there are strong non-resonant second-order nonlinear responses [119, 128, 129] of the interface in the case of some metal(-oxide). Generally, experiments employ visible light and VSFG frequencies far from resonant conditions, therefore, the non-resonant term $\chi^{(2),\mathrm{NR}}$ is approximately off-resonant to the light frequencies involved[130].

Microscopic expression of molecular hyperpolarizability As the electric field is increased, the description of the induced dipole moment μ should include the normally insignificant nonlinear terms. We can express the induced dipole moment as

$$\mu = \mu_0 + \alpha \mathbf{E} + \beta \mathbf{E} \mathbf{E}. \tag{2.38}$$

The VSFG spectra are determined by the frequency-dependent hyperpolarizability in molecular level description. The frequency-dependent hyperpolarizability can be expressed as a sum of resonant and non-resonant terms:

$$\beta_{\eta\xi\kappa}(\omega_{\rm SFG}, \omega_{\rm vis}, \omega) = \beta_{\eta\xi\kappa}^{\rm R} + \beta_{\eta\xi\kappa}^{\rm NR}, \tag{2.39}$$

where η , ξ and κ are space-fixed axes. The resonant term of the frequency-dependent hyperpolarizability is

$$\beta_{\eta\xi\kappa}^{R}(\omega_{SFG}, \omega_{vis}, \omega) = \sum_{v',v} \frac{\langle v | \alpha_{\eta\xi} | v' \rangle \langle v' | \mu_{\kappa} | v \rangle}{(\omega_{v'} - \omega_v) - \omega - i\gamma_{v'v}} \rho_v, \tag{2.40}$$

where the subscripts η , ξ and κ denote body-fixed axes, $\omega_{v'} - \omega_v$ is the vibrational energy gap, ρ_v is the thermal distribution function of the initial vibrational states v, $\alpha_{\eta\xi}$ is the $\eta\xi$ -th component of the molecular dipole polarizability, μ_{κ} is the κ -th component of the molecular dipole moment, and $\gamma_{v'v}$ is the damping rate. Since (Appendix A)

$$\int_0^\infty dt e^{-it[(\omega_{v'} - \omega_v) - \omega - i\gamma_{v'v}]} = \frac{-i}{(\omega_{v'} - \omega_v) - \omega - i\gamma_{v'v}},\tag{2.41}$$

we can rewrite Eq. (2.40) as

$$\beta_{\eta\xi\kappa}^{R} = i \int_{0}^{\infty} dt \sum_{v'v} e^{-i[(\omega_{v'} - \omega_{v}) - \omega - i\gamma_{v'v}]t} \langle v | \alpha_{\eta\xi} | v' \rangle \langle v' | \mu_{\kappa} | v \rangle \rho_{v}$$

$$= i \int_{0}^{\infty} dt \sum_{v'v} e^{i\omega t} \langle v | e^{iHt} \alpha_{\eta\xi} e^{-iHt} | v' \rangle \langle v' | \mu_{\kappa} | v \rangle \rho_{v}$$

$$= i \int_{0}^{\infty} dt e^{i\omega t} \langle \alpha_{\eta\xi}(t) \mu_{\eta\xi}(0) \rangle, \qquad (2.42)$$

where H is the Hamiltonian of the system without external field. Eq. (2.42) indicates that the resonant term $\beta_{\eta\xi\kappa}^{(2),R}$ is the FT of the quantity $\langle \alpha_{\eta\xi}(t)\mu_{\kappa}(0)\rangle$, i.e., the ensemble average of the time correlation function $\alpha(t)\mu_r(0)$. The damping rate $\gamma_{v'v}$ is not explicitly included in Eq. (2.42), because the dephasing is incorporated in the time development of the off-diagonal matrix elements of $\alpha_{\eta\xi}(t)$ and $\mu_{\kappa}(0)$.

The tensor elements $\chi^{(2),R}_{\eta\xi\kappa}$ is microscopically represented as the average sum of first-order hyperpolarizability of the constituent molecules β in the space-fixed frame

$$\chi_{\eta\xi\kappa}^{(2),R} = \langle \sum_{i}^{N} \sum_{pqr} D_{\eta p}(\Omega_{i}) D_{\xi q}(\Omega_{i}) D_{\kappa r}(\Omega_{i}) \beta_{pqr} \rangle$$
 (2.43)

where $D(\Omega_i)$ is the direction cosine matrix of the *i*-th molecule, projecting β onto the space-fixed frame [131].

The Fresnel Factors Because of screening and dipole-dipole coupling, the local electric fields felt by molecules is different from the macroscopic fields[129]. The VSFG signal depends on the magnitude of the local electric fields of the interacting optical beams at the interfaces. While the magnitude of the local electric fields is related to both the intensity of the incident beams and the linear refractive indices of the different layers (bulk) of the sample[122]. The Fresnel coefficients define the magnitude of the electric fields at the interface. Therefore, to find out the magnitude of the local electric fields, we need to evaluate the Fresnel factors. The VSFG intensity $I_{\rm SFG}$, is proportional to the intensities of the incident visible and infrared beams, $I_{\rm vis}$, $I_{\rm color}$, and to the square of the second-order nonlinear susceptibilities, $\chi^{(2)}_{\eta\xi\kappa}(\omega_{\rm SFG})$, of the interface:

$$\chi_{\eta\xi\kappa}^{(2)}(\omega_{\rm SFG}) \propto |\sum_{\eta,\xi,\kappa} L_{\eta\eta}(\omega_{\rm SFG})\chi_{\eta\xi\kappa}^{(2)}(\omega_{\rm SFG})L_{\xi\xi}(\omega_{\rm vis})L_{\kappa\kappa}(\omega)|^2 \sec^2(\theta_{\rm SFG})I_{\rm vis}I \quad (2.44)$$

where η , ξ , κ are the Descartes coordinates of the reference frame; $\omega_{\rm SFG} = \omega_{\rm vis} + \omega$ is the frequency of VSFG beam; $L_{\eta\eta}$, $L_{\xi\xi}$ and $L_{\kappa\kappa}$ are the Fresnel coefficients; $\theta_{\rm SFG}$ is the reflected angle of VSFG beam with respect to the normal direction in the medium.

2.5.2 VSFG spectra from velocity auto-correlation functions

In this paragraph I review the derivation an expression for the calculation of the sum frequency generation spectra of water interfaces that is based on the projection of the atomic velocities on the local normal modes, such an approach permits one to obtain the VSFG signals from suitable velocity ACFs, reducing the computational cost to that of the accumulation of a molecular dynamics trajectory, and therefore cutting the overhead costs associated with the explicit calculation of the dipole and polarizability tensor. Moreover, the method permits to interpret the peaks in the spectrum in terms of local modes. The components of the resonant term $\chi_{\eta\xi\kappa}^{(2),R}$ of the second order susceptibility can be calculated according to the classical formula [22, 130, 132]

$$\chi_{\eta\xi\kappa}^{(2),R} = \frac{-i}{k_{\rm B}T\omega} \int_0^\infty dt e^{i\omega t} \left\langle \dot{A}_{\eta\xi}(t) \dot{M}_{\kappa}(0) \right\rangle \tag{2.45}$$

where $k_{\rm B}$ is the Boltzmann constant, ω is the frequency of the IR beam, \mathbf{M} (A) is the dipole moment (dipole polarizability) of the system, and $\langle \dots \rangle$ denotes the average over all starting time points.

The total dipole moment and dipole polarizability derivatives for the system can be expressed in terms of the water and bond contributions:

$$\dot{A} = \sum_{i=1}^{N} \sum_{\epsilon} \dot{\alpha}^{i,l,\epsilon} \tag{2.46}$$

$$\dot{\mathbf{M}} = \sum_{i=1}^{N} \sum_{\epsilon} \dot{\mu}^{i,l,\epsilon} \tag{2.47}$$

where $\mu^{i,l,\epsilon}$ ($\alpha^{i,l,\epsilon}$) is the dipole moment (polarizability) of the bond ϵ of the *i*-th water molecule, the superscript (l) denote these quantities are measured in the lab frame, and N is the total number of water molecules. Therefore, the correlation function in Eq. (2.45) can be written as

$$\langle \dot{A}_{\eta\xi}(t)\dot{M}_{\kappa}(0)\rangle = \sum_{i=1}^{N} \sum_{\epsilon} \left\langle \dot{\alpha}_{\eta\xi,i,\epsilon}^{l}(t)\dot{\mu}_{\kappa,i,\epsilon}^{l}(0)\right\rangle$$

$$+ \sum_{i=1}^{N} \sum_{\epsilon} \left\langle \dot{\alpha}_{\eta\xi,i,\epsilon}^{l}(t)\dot{\mu}_{\kappa,i,-\epsilon}^{l}(0)\right\rangle$$

$$+ \sum_{i,j=1;i\neq j}^{N} \sum_{\epsilon,\epsilon'} \left\langle \dot{\alpha}_{\eta\xi,i,\epsilon}^{l}(t)\dot{\mu}_{\kappa,i,\epsilon'}^{l}(0)\right\rangle. \tag{2.48}$$

In Eq. (2.48), the first term of the right-hand side is the bond auto-correlation, the second term accounts for the correlation between the two bonds in the same water molecule, and the third them for the correlation between bonds in two different water molecules.

We assume that the bond elongation are small compared to the total bond length and stretching frequencies of the bond are much larger than frequencies of bond reorientation, for example, the libration. Therefore, we can approximately write $\dot{\mu}(0)$ by

$$\dot{\mu}_{\kappa}(0) = \sum_{i}^{x,y,z} \mathbf{D}_{\kappa i}(0)\dot{\mu}_{i}(0)$$

$$= \sum_{i}^{x,y,z} \mathbf{D}_{\kappa i}(0) \left(\sum_{j}^{x,y,z} \frac{d\mu_{i}}{dr_{j}} \frac{dr_{j}}{dt}|_{t=0}\right)$$

$$= \sum_{i}^{x,y,z} \mathbf{D}_{\kappa i}(0) \frac{d\mu_{i}}{dr_{z}} v_{z}(0), \tag{2.49}$$

where $\mathbf{D}_{\kappa i}$ is the direction cosine between the laboratory-fixed κ axis and the molecular-fixed i axis, and $v_z = \frac{dr_z}{dt}|_{t=0}$ is the projection of the velocity on the bond axis.

Similarly, for the dipole polarizability, we have

$$\dot{\alpha}_{\eta\xi}(t) = \sum_{i,j}^{x,y,z} \left(\mathbf{D}_{\eta i}(t) \frac{\partial \alpha_{ij}}{\partial r_z} \mathbf{D}_{\xi j}(t) \right) v_z(t). \tag{2.50}$$

The Eq. (2.49) and Eq. (2.50) simplify the calculation of the $\langle \dot{A}_{\eta\xi}(t)\dot{\mathbf{M}}_{\kappa}(0)\rangle$ in Eq. (2.45), because $v_z(t)$ and $\mathbf{D}(t)$ can be readily determined from the DFTMD trajectory, and $\frac{d\mu_i}{dr_z}$ and $\frac{d\alpha_{ij}}{dr_z}$ can be parameterized[122, 133].

We used three different frameworks: the lab framework (x^l, y^l, z^l) , the molecular

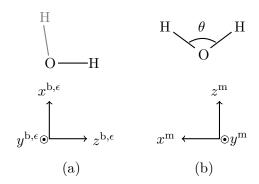


FIGURE 2.2: The representation of the bond (a) and the molecular (b) frameworks. (from Ref. [124])

framework $(x^{\rm m},y^{\rm m},z^{\rm m})$ and the bond framework $(x^{\rm b},y^{\rm b},z^{\rm b})$ (see Fig. 2.2). In the lab framework, the $z^{\rm l}$ -axis is perpendicular to the interface. The molecular frame will be used to decompose the signal into normal modes of water monomers. For the j-th molecule, the $z^{\rm m}$ axis is along the bisector of the H-O-H angle, the $x^{\rm m}$ axis is in the molecular plane, and the $y^{\rm m}$ axis is out of the molecular plane[124].

In the bond framework, $z^{\mathrm{b},\epsilon}$ axis is along the bond ϵ of a molecule, $z^{\mathrm{b},\epsilon}$ is in the molecular plane and $y^{\mathrm{b},\epsilon}$ is out of the molecular plane.

$$\dot{\alpha}^{l,\epsilon} = \mathbf{D}^{m} \mathbf{D}^{b,\epsilon} (\frac{\partial \alpha^{b}}{\partial r} \dot{r}^{\epsilon}) (\mathbf{D}^{b,\epsilon})^{T} (\mathbf{D}^{m})^{T}, \tag{2.51}$$

$$\dot{\mu}^{l,\epsilon} = \mathbf{D}^{\mathrm{m}} \mathbf{D}^{\mathrm{b},\epsilon} (\frac{\partial \mu^{\mathrm{b}}}{\partial r} \dot{r}^{\epsilon}). \tag{2.52}$$

The direction cosine matrix $\mathbf{D}^{b,\epsilon=1}$ and $\mathbf{D}^{b,\epsilon=-1}$ can be expressed as

$$\mathbf{D}^{b,1} = \begin{pmatrix} \cos\frac{\theta}{2} & 0 & -\sin\frac{\theta}{2} \\ 0 & 1 & 0 \\ \sin\frac{\theta}{2} & 0 & \cos\frac{\theta}{2} \end{pmatrix}, \quad \mathbf{D}^{b,-1} = \begin{pmatrix} -\cos\frac{\theta}{2} & 0 & \sin\frac{\theta}{2} \\ 0 & 1 & 0 \\ \sin\frac{\theta}{2} & 0 & \cos\frac{\theta}{2} \end{pmatrix}, \tag{2.53}$$

where θ is the H-O-H angle in a water molecule. We can use \mathbf{D}^{m} to transform the coordinates in a molecular framework to coordinates in the lab framework. Because the orientation of water molecules is changing during the simulation, \mathbf{D}^{m} is time dependent.

Based on the DFTMD simulations, we can implement the parameterization of $\frac{\partial \mu_k}{\partial r_z}$ and $\frac{\partial \alpha_{ij}}{\partial r_z}$ and calculate the correlation function $\langle \dot{A}_{\eta\xi}(t)\dot{M}_{\kappa}(0)\rangle$ through the velocity auto-correlation function. This approximation for estimating the susceptibility retains the details of the interface, including the full electronic structure. And it reduces the computational cost for a total calculation with the instantaneous evaluation of the molecular dipoles and polarizabilities[134]. The parametrization of $\frac{\partial \mu_k}{\partial r_z}$ and $\frac{\partial \alpha_{ij}}{\partial r_z}$ is based on the calculation of Maximally Localized Wannier Functions (MLWF) [135] and can be done through the approach developed by Salanne *et al.*[136] and Khatib *et al.*[124]. The implementation of this parametrization is given in Appendix A.

2.6 Available experimental data

In this chapter, we give the experimental results obtained on salty solutions containing alkali cations and nitrate (iodide) anions[137–139].

From the experimental data of surface tension dependence on solute concentration $d\gamma/dm_2$ at low electrolyte concentrations (≤ 1.5 M)[140–143], the relation of the surface/bulk molar concentration ratio K_p [144] among Li⁺, Na⁺ and K⁺ is:

$$0 = K_{p,Na^{+}} < K_{p,K^{+}} < K_{p,Li^{+}}.$$
(2.54)

i.e., Na^+ is the most surface-excluded in the water solution RNO₃, K^+ is less excluded, and Li^+ is the least excluded cation (see Appendix D.2 for details). In modeling the interfaces of aqueous solutions of alkali metal nitrates, we decided to start with LiNO₃, because the Li^+ ion is the least excluded of the vapor-liquid interface among the alkali metal ions

Hua et al. [139] have recently measured the VSFG spectra of water/vapor interface of LiNO₃ salt solutions in the OH stretching region (3000–3800 cm⁻¹) using Heterodyne Detected VSFG spectroscopy[145–147]. The experimental result of the VSFG intensity of the alkali nitrate interfaces is given by in Fig. 2.3. At a difference with the spectra for the water interface, in the spectra of LiNO₃ solutions, a depletion of the 3200 cm⁻¹ peak is observed, with an enhancement of the 3400 cm⁻¹ peak. A similar behaviour had been observed for the interface of NaNO₃ and Mg(NO₃)₂ solutions [138, 139]. It has been suggested that this depletion of the 3200 cm⁻¹ peak, and in some cases the enhancement of the 3400 cm⁻¹ peak, is an indication that nitrate ions reside at the interface. On the other hand the small cations should have little surface propensity. It has also been argued that the positive electric field found at the interface of NaCl, NaI and NaNO₃ salt solutions is due to the formation of an ionic double layer between anions located near the surface and their counter-cations (e.g. Na⁺) located further below. In Phase-Sensitive (PS) VSFG experiments the magnitude of the induced change in the $\text{Im}\chi^{(2)}$ spectra comparatively to that of the neat water suggested that NO₃ has a surface propensity just in between I⁻ and Cl⁻[148, 149].

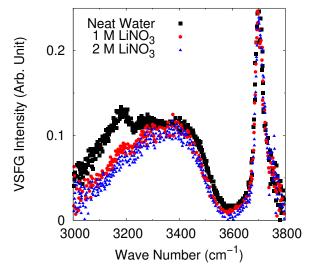


FIGURE 2.3: Experimental VSFG intensity of LiNO₃ solutions, compared with that of neat water[139].

Chapter 3

Alkali nitrate clusters

There are two ways of obtaining highly specific information on solvation structures: studying gas-phase clusters consisting of ions surrounded by a few water molecules [150, 151], and exciting and detecting a dissolved prob molecule [152]. Due to the experimental difficulties, most information on the dynamics of aqueous solvation shells was obtained from MD simulations [153, 154]. In this chapter, DFTMD simulations of gas phase clusters including alkali cations, nitrate ions and a few water molecules have been used to obtain these specific information and to understand the effects of alkali cations and nitrate anions on H-bonding [155, 156]. The VDOS is used to extract the vibrational signatures for the water molecules in these systems. In paragraph 4.1 and 4.2, the effect of the anion and the cation are separately investigated. The two clusters, $[NO_3 \cdot (H_2O)_3]^-$ and $[Li \cdot (H_2O)_4]^+$, are used to study the structural and dynamical properties of water clusters with nitrate ions and with alkali cations. In paragraph 3.3, the effects of alkali metal cations and nitrate anion are discussed within clusters containing both cations and anions and an increasing number of water molecules.

3.1 Cluster of nitrate and water molecules

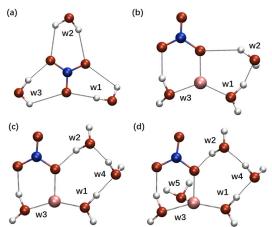


FIGURE 3.1: Geometry optimized structure of the clusters: (a) $[NO_3\cdot(H_2O)_3]^-$; (b) $RNO_3(H_2O)_3$; (c) $RNO_3(H_2O)_4$; (d) $RNO_3(H_2O)_5$ (R=Li,Na,K). More structural properties are given in Appendix \mathbb{C} .

First, we consider the nitrate–water cluster, $[NO_3\cdot(H_2O)_3]^-$. The symmetric isomer of the cluster, as shown in Fig. 3.1 a, is obtained by geometry optimization at the BLYP/TZV2P level of theory. According to the definition of the H-bond[157, 158], there are three H-bonds in it, i.e., only one of the two OH bonds is H-bonded to NO_3^-

in each water molecule. Therefore, the two OH bonds in each water molecule exhibit different vibrational features. Fig. 3.2 shows the difference of the VDOS for the OH bonds in the cluster. For each water molecule, one OH bond is vibrating in the frequency range 3680–3700 cm⁻¹, while the other in the frequency range 3380–3440 cm⁻¹. The difference of frequencies between the vibrational modes is about $\Delta \nu = 180~{\rm cm}^{-1}$.

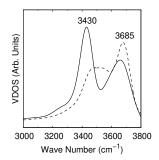


FIGURE 3.2: The VDOS for the two OH bonds in w1 (Fig. 3.1 a) of $[NO_3\cdot(H_2O)_3]^-$.

Additionally, we label the three water molecules as w1, w2, and w3, respectively (Fig. 3.1 a). For the three water molecules, we find some differences in the structural parameters. Table C.5 gives the calculated lengths of H-bonds in the cluster $[NO_3\cdot(H_2O)_3]^-$. The average differences Δd between the H-bonds and the quasi-H-bonds are 0.69 Å (Table C.2).

3.2 Cluster of alkali metal cation and water molecules

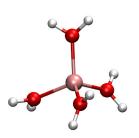


FIGURE 3.3: The cluster $[\text{Li}\cdot(\text{H}_2\text{O})_4]^+$.

To find the effects of alkali cations on the structural properties of water, we investigate the cluster $[\text{Li}\cdot(\text{H}_2\text{O})_4]^+(\text{Fig. 3.3})$. We concentrate on two aspects: the radial distribution function (RDF), and the VDOS for water molecules of this cluster.

The sharp peaks in the RDF given in Fig. 3.4 show that the solvation shell of Li⁺ is bound to all the four water molecules. The peak for g_{LiO} is at 2.02 Å, and for g_{LiH} is at 2.69 Å.

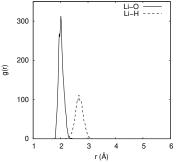
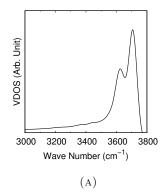


FIGURE 3.4: RDFs $g_{\text{Li-O}}$ and $g_{\text{Li-H}}$ for the cluster $[\text{Li}\cdot(\text{H}_2\text{O})_4]^+$.



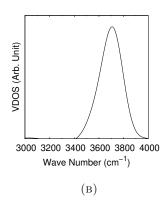


FIGURE 3.5: (A) The VDOS for the four water molecules (all water molecules) in the cluster $[\text{Li}\cdot(\text{H}_2\text{O})_4]^+$. (B) The VDOS for the water molecules (three water molecules) bound to Li in the cluster $[\text{Li}\cdot(\text{H}_2\text{O})_4]^+$.

The VDOS for water molecules in the cluster $[\text{Li}\cdot(\text{H}_2\text{O})_4]^+$ is calculated from a 20-ps trajectory, during which one water molecule escaped from the bonding of the Li and then formed a new HB to another water molecule. First, Fig. 3.5 (A) shows that, in this case, there are two types of OH stretching modes in the cluster $[\text{Li}\cdot(\text{H}_2\text{O})_4]^+$: free OH stretch which peaks at 3705 cm⁻¹ and bonded OH stretch at 3625 cm⁻¹. However, the water molecules just bound to Li has two degenerate free O-H stretching modes. The VDOS for water molecules only bound to Li (Fig. 3.5 (B)) shows that these water molecules only have free OH stretch, since there is only a broad stretching mode at 3705 cm⁻¹.

3.3 Clusters of alkali nitrate and water molecules

As a first minimal model system for the interfaces of alkali nitrate solution, we consider alkali nitrate water clusters including 3 to 5 waters. The idea is to investigate the effect of the alkali nitrate on the vibrational properties of those water molecules which are directly H-bonded to the ions. In our simulations, the clusters are geometry optimized and the most stable configurations are determined (Fig. 3.1 b–d). The first interesting result is that for all the clusters containing 3 to 5 water molecules, a contact ion pair is maintained during the 300 K simulation trajectories where a direct interaction involves the cation and one of the nitrate oxygen's.

In the LiNO₃(H₂O)₃ cluster, there are three H-bonds and three Li-O bonds. The average lengths of them are given in Table 3.1. We use HB1, HB2 and HB3 to denote the HB between w1 and w2, w2 and NO₃⁻, and w3 and NO₃⁻, respectively (Fig. 3.1 b). Both the average lengths of HB1 and HB3 are very close to each other and both of them are smaller than those of HB2. Since both w1 and w2 are bound to Li⁺, we calculate an average value $\bar{d}_{\rm HB} = 1.81$ Å of the lengths of HB1 and HB3. The difference between length of HB2 and $\bar{d}_{\rm HB}$ is $\delta d_{\rm HB} = 0.19$ Å. By testing the difference of environment of each H-bonds, we obtain that $\delta d_{\rm HB}$ comes from the difference between Li-O and H-bonds.

Besides, the RDF between the alkali ($\mathrm{Li^+}$, $\mathrm{Na^+}$ or $\mathrm{K^+}$) and the water O (panel a) and the nitrate O – water H (panel b) are reported in Fig. 3.6. The sharp peaks in the RDF (Fig. 3.6 b) show that the nitrate is solvated and in particular a stronger HB is formed in the presence of the cation.

TABLE 3.1: The average length r_a of H-bonds (Li-O bonds) in the cluster LiNO₃(H₂O)₃.

Bonds	$r_a(\text{ Å})$
HB1	1.83 ± 0.14
HB2	2.00 ± 0.25
HB3	1.79 ± 0.16
O(w1)–Li	1.95 ± 0.09
O(w3)–Li	1.92 ± 0.07
nitrate O–Li	1.91 ± 0.08

The vibrational features associated to the small clusters are calculated from the VDOS and reported in Fig. 3.8. In the frequency range $3000-3800~{\rm cm}^{-1}$, each water molecules has two vibrational bands. In addition to the free OH peak at $3700~{\rm cm}^{-1}$, we see that the HB band is characterized by quite strong red-shifted peaks around $3200~{\rm cm}^{-1}$. These red-shifted peaks are associated to water molecules which are bound either to the cation or to both cation and anion and are different with respect to the peaks associated to the water molecules which only bound to the nitrate in the simple cluster $[{\rm NO}_3 \cdot ({\rm H_2O})_3]^-$ (3430 cm⁻¹, see Fig. 3.2).

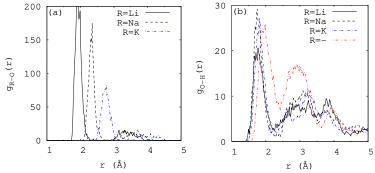


FIGURE 3.6: (a) RDF $g_{\text{R-O}}$ for clusters RNO₃(H₂O)₃ (R=Li, Na, K); (b)RDF $g_{\text{O-H}}$ for clusters RNO₃(H₂O)₃ and [NO₃·(H₂O)₃]⁻ (no alkali metal cation, denoted as "R=-").

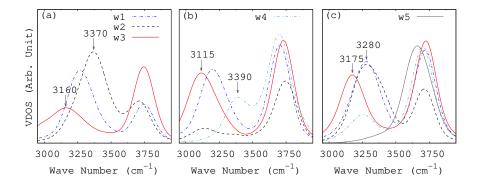


FIGURE 3.7: The VDOS for H_2O in clusters $LiNO_3(H_2O)_n$: (a) n=3; (b) n=4; (c) n=5. w1: H_2O bound to Li and H_2O ; w2: H_2O bound to nitrate and H_2O ; w3: H_2O bound to Li and nitrate; w4: H_2O bound to H_2O ; w5: H_2O only bound to Li.

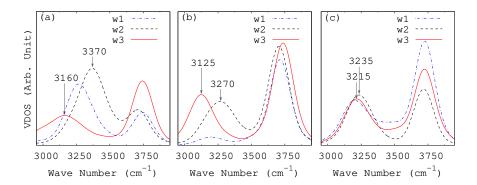


FIGURE 3.8: The VDOS for H_2O in clusters: (a) $LiNO_3(H_2O)_3$, (b) $NaNO_3(H_2O)_3$ and (c) $KNO_3(H_2O)_3$. w1: H_2O bound to R and H_2O ; w2: H_2O bound to nitrate and H_2O ; w3: H_2O bound to R and nitrate.

To explore the effect of adding some additional water molecules to the cluster, we considered the clusters $\text{RNO}_3(\text{H}_2\text{O})_n$ (n=4, 5; R=Li, Na, K). The most stable configurations are shown in Fig. 3.1 c and d, and the corresponding VDOS for water molecules are shown in Fig. 3.7 b and c for the clusters $\text{LiNO}_3(\text{H}_2\text{O})_n$ (n=4 and 5). We find that the OH stretching peaks in the HB region are also quite red-shifted. The red shift is particularly strong for the water molecules which are directly interacting with the Li and those which are simultaneously bound to the Li and to the nitrate O's (e.g. w3).

We also calculate the effects of other alkali metal cations, namely Na⁺ and K⁺. The calculated VDOS for water molecules in clusters NaNO₃(H₂O)₃ and KNO₃(H₂O)₃ are shown in Fig. 3.8 b and c, respectively. As in the case of LiNO₃(H₂O)₃, the HB bands are also characterized by red-shifted peaks around 3200 cm⁻¹. In addition, the peaks in the OH-stretching region are also compatible with infrared predissociation (IRPD) spectra which have been recorded for the [Li·(H₂O)₃₋₄Ar]⁺ clusters [159–161] and for [Na·(H₂O)₄₋₇]⁺ and [K·(H₂O)₄₋₇]⁺ clusters[162], although there no nitrate is present and only the effect of the cation was investigated.

To summarize, the vibrational spectra from the clusters clearly point to red-shifted peaks which are not recorded in the vibrational sum-frequency generation spectra at the water/vapor interface for the ${\rm LiNO_3}$ solution. In other words, these clusters are not really representative of the solvation structures presents in the ${\rm LiNO_3}$ solution. Therefore, these small clusters cannot be directly used to describe the topmost layer of the ${\rm LiNO_3}$ solution, and we need to build more realistic models to capture the main features the interface. In particular, according to the cluster picture one would be tempted to rule out the possibility of a contact ion pair at the interface.

Chapter 4

VSFG spectroscopy of the water/vapor and solution/vapor interfaces

In Chapter 3, we investigated the VDOS for water clusters containing nitrate ions and alkali metal ions. We found that small clusters cannot be directly used to model interfaces of aqueous solutions, and we need to build more realistic ones to capture the main features of interfaces. In this chapter, we will analyze the structure and dynamics of solutions containing an alkali cation and a nitrate (iodide) ion and to provide a microscopic interpretation of recent experimental results[137–139].

The goal of this chapter is to find the origin of the main characteristics of the VSFG spectra of the LiNO $_3$ solution, and provide a molecular picture to interpret the recorded spectra. To achieve this goal, we simulate a solution/vapor interface including Li⁺ and NO $_3^-$, as shown in Fig. 4.1, and extract the vibrational spectroscopic properties of the interface.

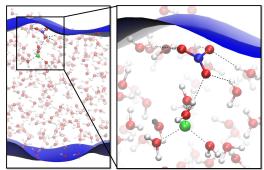


FIGURE 4.1: The salty water interface of LiNO₃ solution (top) and the water/vapor interface (bottom). The right panel shows that the ${\rm Li^+}$ and the ${\rm NO_3^-}$ ions are separated by a water molecule at the salty interface.

We consider a model for the interface where a slab of 256 water molecules containing one Li^+ and one NO_3^- ($\text{LiNO}_3(\text{H}_2\text{O})_{256}$) is included in a periodic simulation box of $19.70 \times 19.70 \times 40.00 \,\text{Å}^3$ at 300 K. The slab is 20 Å thick and infinite in the x and y direction, while the separation between the periodic slabs in the z direction is 20 Å. The LinO_3 was inserted at one of the two interfaces, with the NO_3^- residing in the topmost layer and the Li^+ residing somewhat deeper at about 5 Å from the surface. In this way we have a model with one salty interface and one neat interface which can be used as a reference. To provide the interpretation to the above experimental results, the following analysis tools are used: (1) VDOS; (2) calculation of the nonlinear susceptibility; (3) reconcile of the interface and cluster picture. In paragraph 4.1, the

VSFG spectroscopy of the alkali nitrate interfaces of aqueous solutions is calculated, to find the connection between these two kind of models: the interface and cluster picture. Additionally, to study the effect of cations, the interfaces of alkali-iodine solutions are also studied in paragraph 4.2.

4.1 VSFG spectra of the LiNO₃ solution/vapor interface

It has been often put forward the idea that in nitrate solution anion and cation are paired at the interface and form a double layer. Based on the relatively high propensity of NO_3^- for the interface [63, 163] we decided to start the simulations with the anion at the water surface and to investigate the possibility that LiNO₃ forms a stable water-separated ion pair at the interface. The idea that nitrate anions form water-separated pair where the Coulomb interaction is shielded was already suggested for divalent cation nitrate [63]. The first result is that such model system is stable and the NO_3^- remains within the topmost water layer during all the simulation time. This result can be found in the probability distribution along z-axis of the simulation box, as shown in Fig. 4.2. This is in agreement with previous simulation results based on polarizable classical force field [164] and also with some DFTMD work on nitric acid, which was also found stable at the interface [165]. Moreover, the Li⁺ remains relatively close to the surface, in a water sub-layer forming a water separated ion pair with the NO_3^- at the interface.

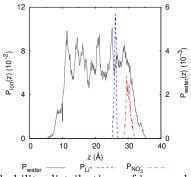


FIGURE 4.2: Probability distributions of ions and water molecules for $LiNO_3$ water interface along the normal direction.

We have calculated the susceptibility for the two interfaces, namely the one containing the LiNO₃ pair(salty interface) and the neat one which does not include any ion. The calculated imaginary part is reported in panel a and the intensity in panel b of Fig. 4.3. The calculated intensity spectra show a depletion of the 3200 cm⁻¹ region as in the experiments. The same feature is also shown in the imaginary part. Also the calculated spectra show that the free OH region is less intense in the salty interface with respect to the water/vapor interface.

To find the microscopic origin of the depression of the lower frequency region, we have also decomposed the salty water interface VDOS into the contributions coming from the different water molecules.

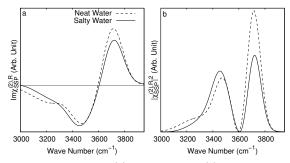


FIGURE 4.3: (a) The $\text{Im}\chi_{SSP}^{(2),R}$ and (b) $|\chi_{SSP}^{(2),R}|^2$ for water molecules at the interface of LiNO₃ solution.

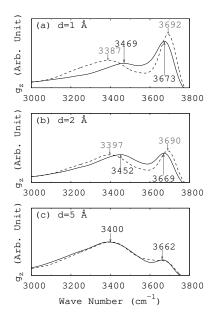


FIGURE 4.4: VDOS $g_z(\nu)$ of water molecules at the interface of LiNO₃ solution (solid line) and at the water/vapor interface (dashed line). (a): d=1 Å; (b): d=2 Å; (c): d=5 Å.

The VDOS $g_z(\nu)$ for the water molecules at the interface, which is calculated from the FT of the auto-correlation function of velocity of the atoms in the z-axis projection, gives a rough value of the thickness of the interface d. Using 1, 2 and 5 Å thicknesses, we have defined three different interfacial regions. For the LiNO₃ solution, $g_z(\nu)$ of the salty and neat water interfaces in the slab is reported in Fig. 4.4. When d=1 Å, water molecules at the solution surface have lower free OH stretching frequency than that in pure water. This means that there are less water molecules with free OH stretch at the interface of LiNO₃ solution than at the water/vapor interface. It compares very well with the experimental result of the surface propensity of nitrate anions in water solution [137]. Meanwhile, compared to the result of the water/vapor interface, the H-bonded band of the VDOS for the salty interface has a blue shift of $\Delta \nu \approx 80~{\rm cm}^{-1}$. As we increase the value of d, the difference between the VDOS of the water/vapor and solution/vapor interface is gradually reduced. For example, when d=2 Å, the amount of blue shift $\Delta\nu$ is reduced to 55 cm⁻¹(Fig. 4.4 b); when d = 5 Å, the amount of blue shift is almost zero (Fig. 4.4 c). This indicates that the ions' (Li⁺, Na⁺, K⁺ and NO₃⁻) effects can be found only on the water molecules in the top \sim 5-Å layer of the interface. As the thickness of the interfacial water layer included in $g_z(\nu)$ increases, the free OH signal is depressed and at the same time the H-bonded OH bands for the salty and neat water interfaces become more similar.

To explore the reason for the blue shift of the H-bonded OH stretch in the interface system, we also calculated the VDOS $g(\nu)$ for the six water molecules in the subsystems $NO_3^-(H_2O)_6$ (The structure of this cluster is shown in Fig. 4.1). Compared to the VDOS for H-bonded water molecules at the water/vapor interface, a blue shift of $\Delta\nu'\approx 80~{\rm cm}^{-1}$ on the vibrational modes of water molecules is found at the interface (Fig. 4.5). It indicates that a HB with nitrate acceptor (nitrate—water bond) is weaker than that with water acceptor (water—water bonds), since the amount of O–H frequency shift reflects the strength of the H-bonds[131, 166]. This feature agrees with experimental result obtained by Jubb et al..[138]. The OH stretching band at 3394 cm⁻¹ (for $T=300~{\rm K}$) also agrees with that of liquid pure water (3400 cm⁻¹[167]). Since the value of $\Delta\nu'$ is almost equal to the value of $\Delta\nu$ at $d=1~{\rm \AA}$, we conclude that the blue shift of the VDOS at the salty water interface is mainly caused by the H-bonds between the uppermost nitrate and water molecules at the solution/vapor interface.

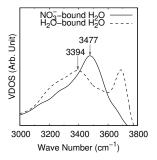


FIGURE 4.5: VDOS for 6 water molecules bound to NO_3^- at the LiNO₃ solution/vapor (LiNO₃/vapor for short) interface (salty water) and that for 15 water molecules at the top layer (d=1 Å) of the neat water.

First, there are two reasons to support the view that NO_3^- is located at the top layer of the surface. (1) The reduced intensity of the free OH peak can be explained by that NO_3^- is at the surface. The 3700 cm⁻¹-peak is the character of free OH stretch in water molecules with their dipole moment pointing to the vapor phase[32, 168]. NO_3^- binds to water molecules from the water surfaces which have less free OH, therefore reduce the intensity of the free OH peak. (2) Those water molecules directly H-bonded to the NO_3^- ion show an higher frequency band with respect to the neat water at the interface, which explain the increased intensity of the 3400 cm⁻¹ band.

Second, the statement that Li⁺ and NO₃⁻ are separated is confirmed by Ref. [169, 170], which show that the alkali metal cations are of small composite partition coefficients $(k_{p,K^+} = 0.00 \pm 0.03, k_{p,Na^+} = 0.05 \pm 0.17, k_{p,Li^+} = 0.14 \pm 0.18)$, i.e., these cations are more surface-excluded than NO₃⁻ $(k_{p,NO_3}^- \approx 1.0)$. How do we reconcile the interface picture and the cluster picture? In the small clusters (with 3, 4 and 5 water molecules) the contact ion pair is the most stable configuration, while at the interface the water separated configuration is the most stable. This suggests that a sufficiently large number of water molecules is required to stabilize a water separated ion pair where the NO₃⁻ anion still reside at the surface. To verify this idea we extracted a relatively large cluster with 30 water molecules from the full interface, centered around the Li⁺ ion and we simulate it in the gas phase. For this medium size cluster we calculated the free energy difference between the water separated and the contact ion pair. The details of the calculation is given in Appendix D.1. The blue-moon ensemble method [171–173] is used to calculate the free energy as a function of a

parameter: the distance r between alkali metal cation and the nitrogen of NO_3^- in $LiNO_3(H_2O)_{30}$. In Fig. 4.6, we find that there are two minima in the free energy at r=2.9 Å (configuration A) and r=4.3 Å (configuration B). Li^+ and NO_3^- are bonded in configuration A, but are water-separated in configuration B. The free energy difference $\Delta F_{AB} = F_A - F_B = 0.3$ kcal/mol. The energy barrier between C and A (B) is: $\Delta F_{CA} = 1.2$ kcal/mol ($\Delta F_{CB} = 1.5$ kcal/mol). Configuration B is more stable than A. For the water molecules in interface system, NO_3^- resides on the surface and Li^+ in the layer below, separated from NO_3^- by water molecules. Therefore, no obvious red-shift induced by alkali metal cation and nitrate is obtained in the VSFG spectrum. Our results show that as the number of waters increases, the first solvation shell around the Li^+ is stabilized and the water separated ion pair is equally stable as the contact ion.

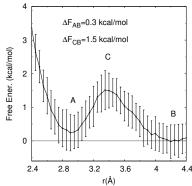


FIGURE 4.6: Free energy profile with respect to the distance r between Li⁺ and the nitrogen in NO₃⁻ in the cluster LiNO₃(H₂O)₃₀. A: configuration A where r = 2.9 Å; B: configuration B where r = 4.3 Å; C: the transition states.

Finally, in the salty interface, one single water is constantly shared between the Li⁺ and the NO₃⁻ and indeed this water shows a vibrational peak with a very pronounced red shift. This clearly reminds the water peak we already observed in the small clusters, however if the full interface is considered its signature do not emerge from the spectra, as it can be seen in Fig. 4.3.

From the three arguments above, our conclusion is thus that in the VSFG spectra we do only see the changes induced by the NO_3^- at the interface. This points to a clear $3400~\rm cm^{-1}$ band in the vibrational spectra. The Li⁺ resides in the sub-layer forming a water separated ion pair at the interface.

Until now, we have analyzed the behaviour of a salty interface containing LiNO₃. Both the measured and calculated VSFG spectra show a reduced intensity of the lower frequency portion of the HB region, namely around 3200 cm⁻¹, when compared to the water/vapor interface. This reduction can be attributed to the H-bonds which are established between the NO_3^- and the surrounding water molecules at the interface. This effect is only related to the presence of NO_3^- at the water surface and is not affected by the presence of Li⁺ ions. Indeed we have shown that although Li⁺ can reside relative close to the water surface, also forming a water mediated ion pair with NO_3^- , its effect on the VSFG spectrum is not visible. The water which mediate the interaction between the NO_3^- and Li⁺ would produce a red-shifted peak in small water cluster, but its influence is not visible neither in the calculated or the measured VSFG spectra. We have also shown that the very simple models, such as small clusters are not suitable to reproduce the experimental spectra and cannot provide a microscopic interpretation of the spectra. A realistic model of the interface is required to address

the perturbation of the ion on the water surface. Besides, how do other ions, such as Li⁺, Na⁺, and K⁺, etc., affect the VSFG spectra of the interface? To test our method and explain the experimental observations, we constructed the interface of lithium iodide, sodium iodide, and potassium iodide solutions and calculated the VSFG spectra. Experiments have proved that iodide ions have similar properties to nitrate ions in many aspects. For example, they tend to interface and are easily polarized. On the other hand, when we analyze the effects of cations, the model containing iodide ions is more simplified than the model containing nitrate ions.

4.2 VSFG spectra of the alkali iodide solution/vapor interfaces

Direct investigations of the dynamics of simple ions, such as I⁻ and Br⁻, at water interfaces, by the x-ray photoelectron spectroscopy [174] and MD simulations [49, 175] have shown that these ions could accumulate at the interface. To provide a molecular interpretation of the recorded spectra we perform here *ab initio* molecular dynamics simulation of salty solutions containing alkali cations and iodine.

A model for the electrolyte solution/vapor interface is built, in which a slab of 118 water molecules containing two Li⁺ cations and two I⁻ anions is included in a period simulation box of $15.60 \times 15.60 \times 31.00$ Å³. This model corresponds to a 0.9 M solution. The slab is about 20 Å thick and infinite in the x and y direction, while the separation between the periodic slabs in the z direction is about 20 Å. In the initial configuration, the LiI was inserted at one of the two interfaces, with the I⁻ residing in the topmost layer and the Li⁺ residing somewhat deeper at about 5 Å from the surface. Using the same method, we also constructed interface models of NaI solution and KI solution for DFTMD simulations. In all the cases the systems were equilibrated for 30 ps and then a production time of 60 ps was considered for the analysis.

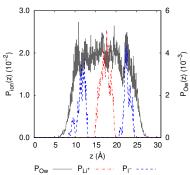


FIGURE 4.7: Probability distributions P(z), along the normal direction (z-axis), of Li⁺, I⁻ and O in LiI solution/vapor interface.

Structural Properties First, we have calculated the probability distributions of Li^+ , I^- and O with respect to the normal direction (z-axis) of the LiI solution surface. The results are given by Fig. 4.7, where we see that I^- anions prefer to be located at the surface of the solution, while the Li^+ cations prefer to stay below the surface. This result is consistent with the calculations from Ishiyama and Morita [51, 54] on a similar system.

The effects of Li⁺ and I⁻ on the organization of water molecules are shown in Li-water (Fig. 4.8 a) and I-water RDFs (Fig. 4.8 b), respectively. In Fig. 4.8 a, the

first two peaks of $g_{\text{Li-O}}$ and $g_{\text{Li-H}}$ are located at 1.97 Å and 4.12 Å, and, 2.61 Å and 4.73 Å, respectively. Here we consider the difference δ_1 between the first peaks' positions of $g_{\text{X-O}}$ and $g_{\text{X-H}}$. Thus, one can determine the differences of the peaks' positions, which are shown in Table 4.1. The difference δ_1 between the first peaks, 0.67 Å, is shorter than the OH bond length R_{OH} in a water molecule which is about 0.98 Å, i.e.,

$$\delta_1 < R_{\text{OH}}.\tag{4.1}$$

This relation reflects that all the water molecules around the Li⁺ have their O atom facing Li⁺. Similarly, we find from Fig. 4.8 b that the distance δ_1 between the first peaks of the two RDFs is 0.93 Å, and it can be seen that δ_1 is slightly equal to $R_{\rm OH}$, i.e.,

$$\delta_1 \approx R_{\rm OH}.$$
 (4.2)

This shows that for the water molecules around the I⁻, only one H atom forms an I–H bond with the I⁻. This relation also implies that I⁻ is essentially at the outermost layer of the solution interface. This is consistent with many of the previous results from MD simulations [46, 176] and experimental results for the increase in surface tension relative to neat water for aqueous solutions of sodium halide salts[47, 177–179].

TABLE 4.1: Peaks of $g_{\text{Li-O}}$ and $g_{\text{Li-H}}$ for the LiI solution. (unit:Å, the same below)

$g_{ m Li-O}$	$g_{ m Li ext{-}H}$	δ_1
1.97	2.64	0.67
4.12	4.73	0.61
6.13	6.93	0.80

Table 4.2: Peaks of $g_{\text{Na-O}}$ and $g_{\text{Na-H}}$ for the NaI solution.

$g_{ m Na-O}$	$g_{ m Na ext{-}H}$	δ_1
2.41	3.02	0.61
4.55	4.96	0.41
6.48	7.20	0.72

Table 4.3: Peaks of g_{K-O} and g_{K-H} for the KI solution.

$g_{ ext{K-O}}$	$g_{ ext{K-H}}$	δ_1
2.84	3.40	0.56
4.71	5.51	0.80
6.78	7.49	0.71

For NaI and KI interfaces, the effects can be seen from Fig. 4.9 and 4.10. For Na⁺ and K⁺, the relation $\delta_1 < R_{\rm OH}$ remains, i.e., $\delta_1 = 0.61$ Å for Na⁺ and $\delta_1 = 0.56$ Å for K⁺. For iodide ions, the relation $\delta_1 \approx R_{\rm OH}$ still holds (See Fig. 4.9 a and 4.10 a, and Table 4.2 and 4.3). For I⁻ in NaI interface, $\delta_1 = 0.96$ Å; and for I⁻ in KI interface, $\delta_1 = 0.94$ Å (See Fig. 4.9 b and 4.10 b). Therefore, these structural properties are similar to that in LiI interface, except the larger solvation shells.

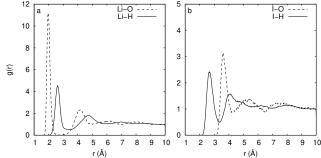


FIGURE 4.8: RDFs for LiI solution/vapor interface: (a) $g_{\text{Li-O}}$ and $g_{\text{Li-H}}$. The first two peaks of $g_{\text{Li-O}}$ and $g_{\text{Li-H}}$: 1.97 and 4.12 Å, and, 2.61 and 4.73 Å, respectively. (b) $g_{\text{I-O}}$ and $g_{\text{I-H}}$. The first two peaks of $g_{\text{I-O}}$ and $g_{\text{I-H}}$: 3.62 and 5.28 Å; and, 2.69 and 4.11 Å, respectively.

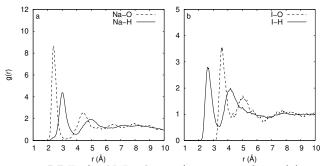


FIGURE 4.9: RDFs for NaI solution/vapor interface: (a) $g_{\text{Na-O}}$ and $g_{\text{Na-H}}$. The first two peaks of $g_{\text{Na-O}}$ and $g_{\text{Na-H}}$: 2.41 and 4.55 Å, and, 3.02 and 4.96 Å, respectively. (b) $g_{\text{I-O}}$ and $g_{\text{I-H}}$. The first two peaks of $g_{\text{I-O}}$ and $g_{\text{I-H}}$: 3.59 and 5.04 Å; and, 2.63 and 4.15 Å, respectively.

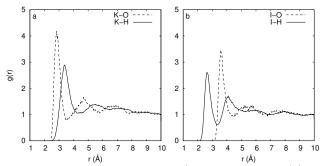


FIGURE 4.10: RDFs for KI solution/vapor interface: (a) $g_{\text{K-O}}$ and $g_{\text{K-H}}$. The first two peaks of $g_{\text{K-O}}$ and $g_{\text{K-H}}$: 2.84 and 4.71 Å, and, 3.40 and 5.51 Å, respectively. (b) $g_{\text{I-O}}$ and $g_{\text{I-H}}$. The first two peaks of $g_{\text{I-O}}$ and $g_{\text{I-H}}$: 3.59 and 5.43 Å; and, 2.65 and 4.10 Å, respectively.

VSFG spectra The calculated VSFG spectra for LiI, NaI and KI interfaces, are shown in Fig. 4.11 to 4.13. In all the cases there is one free OH stretching band (3600–3800 cm⁻¹) and one bonded OH stretching band (3000–3600 cm⁻¹). These features are the same as that of the water/vapor interface. The sign of $\text{Im}\chi^{(2),R}$ is positive for the free OH peak while it is negative in the hydrogen bonded region. This result is consistent with the VSFG spectrum calculated in paragraph 4.1, i.e., (1) the anion–water hydrogen bonds at interfaces decrease the amount of free stretching OH bonds (2) the free stretching peak in the intensity of VSFG decrease and the H-bonded stretching peak is shifted at the interfaces of LiNO₃ (or alkali-iodine) solution.

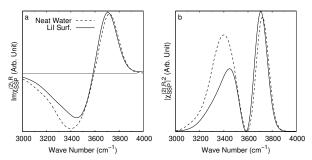


FIGURE 4.11: (a) $\text{Im}\chi_{SSP}^{(2),R}$ and (b) $|\chi_{SSP}^{(2),R}|^2$ of the LiI solution/vapor (solid line) and the water/vapor (dashed line) interface. The data for the water/vapor interface is calculated from the DFTMD simulation for the water interface with the same thickness (5 Å) (the same below).

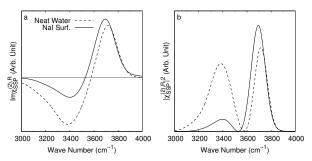


FIGURE 4.12: (a) $\text{Im}\chi_{SSP}^{(2),R}$ and (b) $|\chi_{SSP}^{(2),R}|^2$ of the NaI solution/vapor (solid line) and the water/vapor (dashed line) interface.

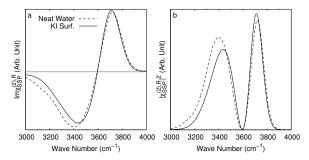


FIGURE 4.13: (a) $\text{Im}\chi_{SSP}^{(2),R}$ and (b) $|\chi_{SSP}^{(2),R}|^2$ of the KI solution/vapor (solid line) and the water/vapor (dashed line) interface.

Compared with the water/vapor interface, the OH-bonded peak of $|\chi_{SSP}^{(2),R}|^2$ for the NaI solution is blue-shifted, which is consistent with experimental results on the NaI solution [40, 48, 138, 180]. The H-bonded OH-stretching peak of $|\chi_{SSP}^{(2),R}|^2$ for LiI and KI solutions are also blue-shifted. These results support the idea that I⁻ is a strong structure-breaking anion. Secondly, the bonded OH-stretching region of the NaI solution is narrower than that of pure water. This result has also been obtained for the interface of LiNO₃ solution. The retained high frequencies of these bonded OH-stretching peaks indicate that these molecules at the interfaces of these solutions are participating in weak H-bonding. The introduction of I⁻ salts, caused a slight decrease in the strong H-bonding region at 3200 cm⁻¹ and relatively an increase in the weak H-bonding region at 3400 cm⁻¹. This result is consistent with experimental results in Ref. [138, 180].

Because of surface isotropy of the solutions[181], the $\chi_{SSP}^{(2),R}$ can be calculated either through $\chi_{XXZ}^{(2),R}$, or $\chi_{YYZ}^{(2),R}$. In our simulation, both of them give very similar results. Here we report the comparison between $\text{Im}\chi_{XXZ}^{(2),R}$ and $\text{Im}\chi_{YYZ}^{(2),R}$ for the KI solution in Fig. 4.14. They should be very close to each other, because the interfaces have rotational symmetry about the z-axis. It can be seen that the calculated spectra are very close to each other. From the results of the nonlinear susceptibilities, we conclude that water molecules at the interfaces of the LiI, NaI, and KI solutions are participating in weaker H-bonds, compared with those at the water/vapor interface. The simulation results permit to interpret the features present in the experimental spectra, which can be explained as consequence of the double layer formed by I⁻ ions on the topmost water layer and the alkali in the sublayer (bulk in our relatively small simulation box). Therefore, the VSFG experimental characteristics on alkali nitrate and alkali iodide solutions can be reproduced from our calculation method, which verifies the reliability of the method used in the current thesis to calculate SFG.

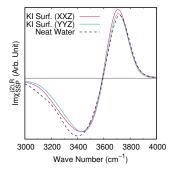


FIGURE 4.14: $\text{Im}\chi_{XXZ}^{(2),R}$ and $\text{Im}\chi_{YYZ}^{(2),R}$ spectra for the KNO₃ solution/vapor interface.

In view of the close relationship between the HB network and the VSFG spectra of the solution interface, we will continue to study the properties of the water/vapor interface and the solution/vapor interface from the perspectives of HB dynamics and reorientation of water molecules in Chapters 5 and 6.

Chapter 5

Hydrogen bond dynamics at the water/vapor interface

Hydrogen bonds play a critical role in the behaviour of bulk water [66, 182, 183], water near interfaces, [184] and aqueous solutions [185]. There are many methods to study the HB dynamics in water, solutions or interfaces, such as molecular dynamics simulation, [184, 186–189] neutron scattering, IR spectroscopy [190, 191], etc. In this chapter, we will introduce the general concepts and methods of HB dynamics [182, 192, 193] used to analyze the structure and dynamic properties of bulk water and the water/vapor interface.

5.1 Definitions of HB population and correlation functions

Luzar and Chandler [192] have pioneered the analysis of the HB dynamics of pure water, and subsequently such analysis has been also extended to more complex systems, e.g., electrolytes[154], protein and micellar surfaces[194]. There are temporal, geometric[195] and energetic criteria [196] to define HB. Here we use the geometric one. Two water molecules are H-bonded if their interoxygen distance between of specific tagged pair of water molecules is less than cutoff radius r_{OO}^c and the O-H···O angle is less than cutoff angle $\phi^{c}[157, 158, 197]$. The value r_{OO}^{c} corresponds to the firstminimum position of the O-O Radial Distribution Function (RDF) of water. [196] The choice for the cutoff angle ϕ^{c} for water-water molecules is obtained by studying the average number of H-bonds, as a function of ϕ^{c} [198]. We call this definition of HB the Acceptor-Donor-Hydrogen (ADH) criterion. To compare the impact of different HB definitions on HB dynamics, we also use another definition of HB in our analysis. When the distance between the oxygen atoms of two water molecules is less than $r_{\rm OO}^{\rm c}$, and the oxygen-hydrogen-oxygen included angle is greater than cutoff angle $\theta^{\rm c}$, then we say that there is a HB between the two molecules. We denote this definition as the Acceptor-Hydrogen-Donor (AHD) criterion. In this thesis, we use $r_{OO}^{c} = 3.5 \text{Å}$ both for ADH and AHD criteria, $\phi^{\rm c}=30^{\circ}$ for ADH criterion, and $\theta^{\rm c}=120^{\circ}$ for AHD criterion.

The configuration criterion above allows us to define a variable h[r(t)] = h(t), the HB population. Here an instantaneous configuration r(t) denotes the positions of all the atoms in the system at time t[192]. The h(t) has a value 1 when the particular tagged pair of molecules are bonded, and 0 otherwise. The fluctuation or deviation in a dynamical variable A(t) from its time-independent equilibrium average $\langle A \rangle$, is defined by [193]

$$\delta A = A(t) - \langle A \rangle.$$

Since the probability that a specific pair of molecules is bonded in a large system is extremely small, i.e., the time average of h is zero, or $\langle h \rangle = 0$, then

$$\delta h(t) = h(t).$$

Therefore, h(t) describe the fluctuation $\delta h(t)$ of the HB population.

While the equilibrium average of $\delta h(t)$ is zero, we can obtain useful information by considering the equilibrium correlations between fluctuations at different times. The correlation between $\delta h(t)$ and $\delta h(0)$ can be written as

$$\langle \delta h(0)\delta h(t)\rangle = \langle h(0)h(t)\rangle - \langle h\rangle^2 = \langle h(0)h(t)\rangle,$$

where the averaging $\langle \cdots \rangle$ is to be performed over the ensemble of initial conditions.

In this paragraph, we will use the following three correlation functions to describe the HB dynamics of the water/vapor interface: the HB population auto-correlation function $C_{\text{HB}}(t)$, the continuum HB population correlation function (survival probability) $S_{\text{HB}}(t)$ and the reactive flux k(t)[199].

HB population auto-correlation function $C_{HB}(t)$ We use the auto-correlation function $C_{HB}(t)$ (c(t) for short) of the HB population to describe the structural relaxation of H-bonds:

$$C_{\rm HB}(t) = \langle h(0)h(t)\rangle/\langle h\rangle. \tag{5.1}$$

With the aid of the ergodic principle, the ensemble average $\langle \cdots \rangle$ is implemented by time average. The $\langle h \rangle$ is the probability that a pair of randomly chosen water molecules in the system is H-bonded at any time t. As examples, the dynamics of the interoxygen distance $r_{\rm OO}(t)$, the cosine of H-O···O angle $\cos\phi(t)$ and the h(t) for a HB in a DFTMD simulated water cluster $({\rm H_2O})_n$ (n=5) at 300 K is displayed in Fig. 5.1, respectively.

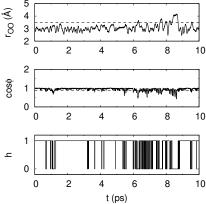


FIGURE 5.1: Dynamics of $r_{OO}(t)$ (top panel), $\cos\phi(t)$ (middle panel), and h(t) (bottom panel) for a typical HB in a water cluster. The dashed lines show the interoxygen distance boundary $r_{OO}^c=3.5$ Å (top panel) and criterion of cosine of $H-O\cdots O$ angle $\cos\phi^c$ with $\phi^c=30^\circ$ (middle panel), respectively.

In a large system that consist of many water molecules, the probability that a specific pair of water molecules are H-bonded is extremely small. Therefore, the $C_{\rm HB}(t)$ relaxes to zero, when t is large enough. The $C_{\rm HB}(t)$ measures correlation in h(t) independent of any possible bond breaking events. This function is similar to

one of the intermittent HB correlation functions, introduced by Rapaport[199], and can be studied by a continuous function, probability densities. From $C_{\rm HB}(t)$, the HB relaxation time can be computed by

$$\tau_{\rm R} = \frac{\int t C_{\rm HB}(t) dt}{\int C_{\rm HB}(t) dt}.$$
 (5.2)

The $C_{\rm HB}(t)$ for the simulated bulk water is shown in Fig. 5.3 (For computational details, see Appendix B). We can obtain the relaxation time from Eq. 5.2: $\tau_R = 14.01$ ps for ADH definition, and $\tau_R = 14.16$ ps for AHD definition.

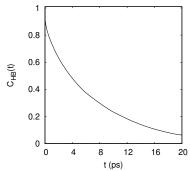


FIGURE 5.2: Time dependence of $C_{\rm HB}(t)$ for bulk water.

The method of water-water pair based HB dynamics used in this section has been frequently used in previous literature [154, 192, 200]. The basis is the population operator h(t) of the HB formed between two water molecules. The function $C_{\rm HB}(t)$ is interpreted as the probability that the HB between a certain pair of water molecules is intact at time t, if the pair of water molecules are H-bonded at time zero.

Besides h(t), here we also discuss the definition of another possible definition of the HB population operator h'(t). When a pair of water molecules a and b are H-bonded, the oxygen atom in each water molecule can act both as a donor and an acceptor. In particular, a pair of water molecules can form 4 different forms of H-bonds. In other words, if the role of H atoms between the pair of water molecules changes, but they still form H-bonds, we think that an old HB is broken and a new HB is formed. The calculated results of the correlation functions (still denoted by $C_{\rm HB}(t)$) of h'(t) of water molecules in bulk water is shown in Fig. 5.3. Comparing Fig. 5.2 and Fig. 5.3, we found that there are some differences in the correlation functions of the two definitions h(t) and h(t), because hydrogen exchange is considered in the O-H pair-based HB population h'(t), but not in the water-water pair-based HB population h(t). In Fig. 5.3, we plotted $C_{\rm HB}(t)$ for bulk water, using ADH and AHD criteria of HB. We found that the difference from different choice of HB population is not affected by the HB criteria. In the following, we will use the HB population h(t) based on molecular pairs, i.e., water-water molecule pairs, or ion-water molecule pairs, unless otherwise specified.

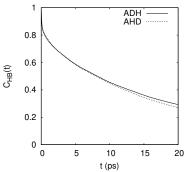


FIGURE 5.3: The $C_{\rm HB}(t)$ for bulk water, as computed from the ADH (solid line) and AHD (dashed line) criterion of H-bonds. Ref:[201]

Because the thermal motion can cause distortions of H-bonds from the perfectly tetrahedral configuration, water molecules show a librational motion on a time scale of ~ 0.1 ps superimposed to rotational and diffusional motions (> 1 ps), which causes a time variation of interaction parameters. A new HB population $h^{(d)}(t)$ was also defined to obviate the distortion of real HB dynamics due to the above geometric definition[154, 196]. The $h^{(d)}(t)$ is 1 when the interoxygen distance of a particular tagged pair of water molecules is less than $r_{\rm OO}^c = 3.5$ Å at time t and 0 otherwise. The difference between the operators $h^{(d)}(t)$ and h(t) is that those molecular pairs that meet the condition of $h^{(d)}(t) = 1$ may not meet the condition of $h^{(d)}(t) = 1$ may have been broken, but they may more easily form H-bonds again. The function

$$C_{\rm HB}^{(d)}(t) = \langle h(0)h^{(d)}(t)\rangle/\langle h\rangle \tag{5.3}$$

is the probability that the specific two water molecules are located in reformable region $(r_{\rm OO} < r_{\rm OO}^{\rm c})$ at time t, if they were H-bonded at time zero. The correlation function

$$n(t) = \langle h(0)[1 - h(t)]h^{(d)}(t)\rangle/\langle h\rangle$$
(5.4)

represents the probability at time t that a tagged pair of initially H-bonded water molecules are unbonded but remain separated by less than r_{OO}^c . In the above formula, 1 - h(t) describes the breaking of a HB at time t after its formation at time t = 0.

The probability at time t that a pair of water molecules bonded by H-bonds at the initial moment does not be bonded but the distance between their oxygen atoms is still less than r_{OO}^c is calculated according to

$$n(t) = \int_0^t dt' k_{\rm in}(t'), \tag{5.5}$$

where $k_{\rm in}(t) = -\langle \dot{h}(0)[1-h(t)]h^d(t)\rangle/\langle h\rangle$ is the restricted rate function.

Continuum HB population correlation function $S_{HB}(t)$ Another scheme to describe the HB dynamics is the continuum HB population correlation function $S_{HB}(t)$ (s(t) for short), or survival probability [154] for a newly generated HB. It is defined as

$$S_{\rm HB}(t) = \langle h(0)H(t)\rangle/\langle h\rangle,$$
 (5.6)

where H(t)=1 if the tagged pair of molecules, remains *continuously* H-bonded till time t and 0 otherwise. It describes the probability that an initially H-bonded molecular pair remains bonded at all times up to t[202]. The $S_{\rm HB}(t)$ for the simulated bulk water according to the formula 5.6 is shown in Fig. 5.4.

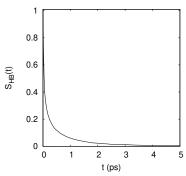


FIGURE 5.4: Time dependence of $S_{\rm HB}(t)$ for bulk water.

The average continuum HB lifetime $\langle \tau_a \rangle$ is calculated by the integration of $S_{\text{HB}}(t)$ over t (For detailed derivation, see Appendix F.1.):

$$\langle \tau_{\rm a} \rangle = \int_0^\infty dt S_{\rm HB}(t).$$
 (5.7)

The time derivative of $S_{\rm HB}(t)$

$$P_a(t) = -\frac{\mathrm{d}S_{\mathrm{HB}}(t)}{\mathrm{d}t} \tag{5.8}$$

represents the first passage time probability density of H bonds. $P_a(t)$ is also called probability distribution of HB lifetimes [203–206], or histogram of HB lifetimes [207, 208]. It denotes the probability of the first HB breaking in time t after it has been detected at t = 0, i.e.,

$$S_{\rm HB}(t) = \int_t^\infty P_a(t')dt'. \tag{5.9}$$

Reactive Flux k(t) The rate of relaxation to equilibrium is characterized by the reactive flux correlation function,

$$k(t) = -\frac{\mathrm{d}C_{\mathrm{HB}}(t)}{\mathrm{d}t},\tag{5.10}$$

i.e., $\langle j(0)[1-h(t)]\rangle/\langle h\rangle$, where $j(0)=-\mathrm{d}h/\mathrm{d}t|_{t=0}$ is the integrated flux departing the HB configuration space at time t=0 (For detailed derivation, see Appendix F.2.). The reactive flux k(t) quantifies the rate that an initially present HB breaks at time t, independent of possible breaking and reforming events in the interval from 0 to t. Therefore, k(t) measures the effective decay rate of an initial set of H-bonds[193, 209].

We assume that each HB acts independently of other H-bonds[192, 210], and due to detailed balance condition, we can obtain

$$\tau_{\rm HB} = \frac{1 - \langle h \rangle}{k},\tag{5.11}$$

where k is the rate constant of breaking a HB (forward rate constant)[211, 212]. For an aqueous interface, the probability of exactly a tagged molecule pair forming a HB is very low, that is, $\langle h \rangle \ll 1$. Therefore, k is related to the average HB lifetime by $\tau_{\rm HB} = 1/k$. We use k' to represent the rate constant from the HB on state to the HB off state for a tagged pair of molecules (backward rate constant). Therefore, the reaction time constant $\tau_{\rm re}$ is

$$\tau_{\rm re} = \frac{1}{k+k'}. \tag{5.12}$$

5.2 Dynamical properties of H-bonds in bulk water and at the water/vapor interface

The bulk water system and the interface between pure water and vacuum, i.e., the water/vapor interface, are ideal model systems for testing our algorithms. For bulk water, we can compare the results of the method in this paragraph with that of previous works[192, 213]. After the validation for bulk water, we will show in this paragraph the results of the HB dynamics of the water/vapor interface.

All simulations in this chapter were performed at 300 K within the canonical (NVT) ensemble. The length of the trajectory is 60 ps of physical time. The simulated bulk water consisted of 128 water molecules in a periodic cubic box of length L = 15.64 Å, which corresponds to a density of 1.00 g cm⁻³. The simulated water/vapor interface consisted of 128 water molecules in a periodic box with size $15.64 \times 15.64 \times 31.28 \text{ Å}^3$.

Correlation functions c(t), n(t) and k(t) First, we calculated the basic structural properties of the simulated bulk water. The RDFs $g_{OO}(r)$ and $g_{OH}(r)$ for bulk water system are shown in Fig. 5.5. The correlation functions $C_{HB}(t)$ from the trajectory

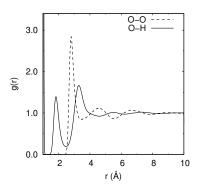


FIGURE 5.5: Partial RDFs of the simulated bulk water.

of a simulation calculated according to formula 5.1 with ADH (solid line) and AHD (dashed line) definition of H-bonds are shown in Fig. 5.6 a. The reactive flux k(t) calculated according to formula 5.10 (see Fig. 5.6 b) is very consistent with the result in [214]. For bulk water, there exists a ~ 0.2 -ps transient period, during which k(t) quickly changes from its initial value[209]. However, at longer times, the k(t) is independent of the HB definitions. The calculation results in Fig. 5.6 b show that when t is large enough, the difference in k(t) caused by different HB definitions is relatively small. Therefore, the long time decay of k(t) reflects the general properties of H-bonds, and calculating the reactive flux HB correlation functions is a more rigorous way to obtain the nature of H-bonds[210].

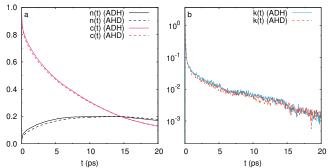


FIGURE 5.6: Time dependence of (a) n(t), c(t) and (b) k(t) for bulk water.

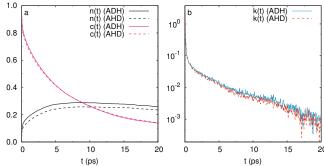


FIGURE 5.7: Time dependence of (a) n(t), c(t) and (b) k(t) for the water/vapor *interface*.

Let's now discuss the result for the water/vapor interface. We reported correlation functions c(t) and n(t) in Fig. 5.7 a and the reactive flux k(t) in Fig. 5.7 b. In both bulk and interface cases, k(t) quickly changes from its initial value on a time scale of less than 0.2 ps. This value can be seen from Fig. 5.8, which redraws the k(t) in Fig. 5.6 and Fig. 5.7 in double logarithmic coordinates and compares them.

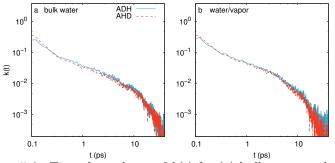


FIGURE 5.8: Time dependence of k(t) for (a) bulk water and (b) the water/vapor interface.

For the water/vapor interface, we focus on the reactive flux k(t), which had been used in the study of HB dynamics of liquid water[192, 201]. The k(t) calculated from the trajectory of water molecules in simulations, is reported in Fig. 5.9. In the water/vapor interface, k(t) quickly changes from its initial value in transient period (~ 0.2 ps). Beyond this transient period, k(t) decays to zero monotonically, and the slope of the $\ln k(t)$ increases monotonically with t (see Fig. 5.9). These two properties have been found for bulk water using the SPC water model by Luzar and Chandler[192]. This log-log plot of the k(t) shows that, as in bulk water, this decay behaviour does not coincide with a power-law decay for the water/vapor interface.

This result is also in good agreement with that of the classical molecular simulation of pure water [182, 214].

The functions n(t) calculated according to Eq. 5.5 for bulk water and the water/vapor interface are shown in Fig. 5.10. It shows that the overall trend of n(t) does not depend on the choice of HB definition. i.e., as t increases, n(t) increases rapidly from 0, and it reaches a maximum at $t \approx 10$ ps, and then gradually decreases. We also find that the maximum of n(t) for the water/vapor interface is slightly higher than that in bulk water. It can be seen that n(t) for the water/vapor interface is always greater than that in bulk water, whether we take the definition of ADH or AHD. We interpret this result as the fact that at time t, there is a greater probability that H-bonds at the interface are broken compared to H-bonds in bulk water.

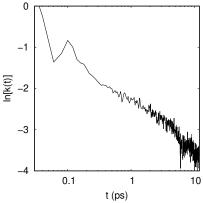


FIGURE 5.9: Time dependence of k(t) for the water/vapor interface, according to Eq. 5.10.

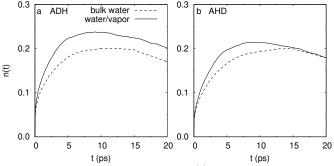


FIGURE 5.10: Time dependence of n(t) for bulk water and the water/vapor interface from (a) the ADH and (b) the AHD criteria.

Reaction rate constants k and k' Khaliullin and Kühne have studied the H-bonding kinetics of bulk water using the AIMD simulations[201]. Based on the HB population operators h(t) and $h^{(d)}(t)$, and correlation functions n(t) and k(t), they have used the simulation data to obtain the ratio k/k' in bulk water, and then the lifetime and relaxation time of the HB. Here, we use the DFTMD simulations to study the HB dynamics at the water/vapor interface. We can obtain the optimal solution range of k and k' from the relationship between the reactive flux and the HB population correlation function c(t) and n(t), and the two rate constants k and k', i.e.,

$$k(t) = kc(t) - k'n(t).$$
 (5.13)

We found the optimal value of the rate constants, k and k', by a least squares fit of the calculated data k(t), c(t) and n(t) beyond the transition phase. The function c(t) was regarded as a P-dimensional column vector composed by $(c(1), c(2), \dots, c(P))^T$, and denoted as \mathbf{c} , with c(i) representing the value of the correlation c(t) at t = i. Similarly, the functions n(t) and k(t) were also viewed as P-dimensional column vectors and are denoted as \mathbf{n} and \mathbf{k} , respectively. Therefore, k and k' were determined from the matrix $\mathbf{A} = \begin{bmatrix} \mathbf{c} & \mathbf{n} \end{bmatrix}$, i.e.,

$$\begin{bmatrix} k \\ -k' \end{bmatrix} = (\mathbf{A}^T \mathbf{A})^{-1} \mathbf{A}^T \mathbf{k}. \tag{5.14}$$

For bulk water and the water/vapor interface, the optimal k and k' are reported in Table 5.1 and 5.2.

TABLE 5.1: The k and k' for bulk water (bulk) and the water/vapor interface (w/v). We carried on the short time region 0.2 ps < t < 2 ps. The unit for k (k') is ps⁻¹, and that for $\tau_{\rm HB}$ (= 1/k) is ps. (Same below.)

Criterion	k (bulk)	k' (bulk)	$ au_{ m HB}~(m bulk)$	k (w/v)	k' (w/v)	$ au_{ m HB}~({ m w/v})$
ADH	0.30	0.99	3.38	0.32	0.77	3.10
AHD	0.29	1.15	3.47	0.31	0.89	3.18

TABLE 5.2: The k and k' for bulk water (bulk) and the water/vapor interface (w/v). We carried on the long time region 2 ps < t < 12 ps.

Criterion	k (bulk)	k' (bulk)	$ au_{ m HB}~(m bulk)$	k (w/v)	k' (w/v)	$ au_{ m HB}~({ m w/v})$	
ADH	0.12	0.04	8.72	0.16	0.07	6.37	
AHD	0.11	0.05	9.50	0.16	0.09	6.47	

To obtain the forward and backward rate constants (k and k'), we performed the fitting in different time region 0.2 < t < 2 ps and 2 < t < 12 ps, respectively. We note that in the larger time region, i.e., 2 < t < 12 ps, the value of HB lifetime τ_{HB} is larger than that in shorter time region, 0.2 < t < 2 ps, no matter for bulk water or for the water/vapor interface. A larger τ_{re} value means that the distance between a water molecule and another water molecule stays within $r_{\text{OO}}^c = 3.5$ Å for a longer time. For the long time region, these values of the k are comparable in magnitude to that obtained by Ref. [201].

5.3 Instantaneous interfacial HB dynamics

To study the HB dynamics for the water/vapor interface, we first determine the instantaneous interface and then define the interfacial HB population operator. Based on these two definitions, we derived the correlation functions and reaction rate constants, for interfacial layers. Using these quantities we can discuss the change in the HB dynamics at the interface with the interface layer's thickness.

Instantaneous Interfaces As Willard and Chandler mentioned, due to molecular motions, interfacial configurations change with time, and the identity of molecules that lie at the interface also change with time, generally useful procedures for identifying interfaces must take into account these motions[74]. To determine the instantaneous interface of the system, we adopted their proposed method based on spatial

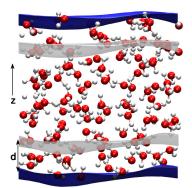


FIGURE 5.11: A slab of water (128 water molecules are included) with the instantaneous interface represented as a blue mesh on the upper and lower phase boundary. The normal is along the z-axis and the parameter d is the thickness of the interfacial layer. The grey surfaces are obtained by translating the interfaces to the interior of the slab along the z-axis (or the opposite direction) by d.

density. The coarse-grained density at space-time point (\mathbf{r},t) can be expressed as polynomial

$$\bar{\rho}(\mathbf{r},t) = \sum_{i} \phi(|\mathbf{r} - \mathbf{r}_{i}(t)|;\xi)$$
(5.15)

where $\mathbf{r}_i(t)$ is the position of the *i*th particle at time *t* and the sum is over all such particles, and

$$\phi(\mathbf{r};\xi) = (2\pi\xi^2)^{-3/2} \exp(-r^2/2\xi^2)$$
(5.16)

is a normalized Gaussian functions for a 3-dimensional system, where r is the magnitude of \mathbf{r} , and ξ is the coarse-graining length. Equation 5.16 is introduced to improve the accuracy of the interface, such that we can extend the domain and make it a single unicom, i.e., no cavity exists in the domain. With the parameter ξ set, the interfaces can be defined to be the 2-dimensional manifold $\mathbf{r} = \mathbf{s}$ such that

$$\bar{\rho}(\mathbf{s};t) = \rho_c \tag{5.17}$$

where ρ_c is a reference density. This interface is a function of time as molecular configurations changes with time, that is $\mathbf{s}(t) = \mathbf{s}(\{\mathbf{r}_i(t)\})$.

After the surface (instantaneous interface) is defined, we can define an instantaneous interface layer for any non-uniform fluid system. Specifically, for the simulated water/vapor interface system in the cuboid simulation box, we can get another two-dimensional manifold $\mathbf{s}_0(t)$ by moving the surface $\mathbf{s}(t)$ determined above along the system's normal coordinate to a certain distance d (two grey surfaces are shown in Fig. 5.11). We use these two surfaces as the two boundaries of the interface. In other words, at any time point t, the volume between the two surfaces $\mathbf{s}(t)$ and $\mathbf{s}_0(t)$ is defined as an instantaneous interface, or instantaneous interface layer. Here, we use d to denote the thickness of the interface. As we change the value of d, we can get interfaces with different thicknesses. Different values of d give us different layering strategies for the interface system. See Fig. 5.11 as an example.

Below we will combine the instantaneous interface and Luzar-Chandler's HB population operator [192] to select the H-bonds at the water/vapor interface. The dynamics of these H-bonds will vary with the thickness d of the interface. By investigating the characteristics of HB dynamics for these interfaces, we can obtain the dynamical

characteristics of various solution/vapor interfaces.

Interfacial hydrogen bond population Once we have determined the instantaneous surface $\mathbf{s}(t) = \mathbf{s}(\{r_i(t)\})$, we can define *interfacial H-bonds*. Now we define the interface HB population operator $h^{(\mathbf{s})}[r(t)]$ as follows: It has a value 1 when the particular tagged molecular pair i, j are H-bonded, and both molecules are inside the instantaneous interface with a thickness d, and zero otherwise:

$$h^{(s)}[r(t)] = \begin{cases} 1 & i, j \text{ are H-bonded, and} \\ & i, j \text{ are inside the interfacial layer} \\ 0 & \text{otherwise} \end{cases}$$
 (5.18)

The definition of $h^{(s)}[r(t)]$ is critical to help us to efficiently obtain the H-bonds' dynamic characteristics of the interfacial layer with a given thickness d. In this paragraph, we discuss H-bonds based on water molecule pairs.

Similar to the correlation function $C_{\rm HB}(t)$ in Eq. 5.1, which describes the fluctuation of the general H-bonds, we define the correlation function $C_{\rm HB}^{({\rm s})}(t)$ that describes the fluctuation of the interfacial H-bonds:

$$C_{\rm HB}^{(s)}(t) = \langle h^{(s)}(0)h^{(s)}(t)\rangle/\langle h^{(s)}\rangle.$$
 (5.19)

Similarly, we define correlation functions

$$n^{(s)}(t) = \langle h^{(s)}(0)[1 - h^s(t)]h^{(d,s)} \rangle / \langle h^{(s)} \rangle, \tag{5.20}$$

and

$$k^{(s)}(t) = -\frac{dC_{\text{HB}}^{(s)}}{dt}.$$
 (5.21)

Using these new correlation functions, we can determine the reaction rate constant of breaking and reforming and the lifetimes of interfacial H-bonding. We will discuss the dependence of the correlation functions $C_{\rm HB}(t)$, $C_{\rm HB}^{(\rm s)}(t)$, and the reaction rates k and k' on the interface thickness d in the next two paragraphs.

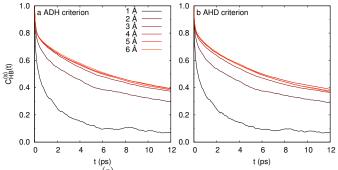


FIGURE 5.12: The $C_{\rm HB}^{(\rm s)}(t)$ for the instantaneous interfacial H-bonds with different thickness (d), as computed from (a) the ADH and (b) the AHD criteria.

Depth-dependence of $C_{\rm HB}^{({\bf s})}(t)$ For the water/vapor interface, we used two geometric criteria of H-bonds to calculate the $h^{({\bf s})}(t)$ and therefore correlation function $C_{\rm HB}^{({\bf s})}(t)$. The calculation results of the $C_{\rm HB}^{({\bf s})}(t)$ are shown in Fig. 5.12. We find that

the HB dynamics is faster at the water/vapor interface when compared to bulk. As d increases, we get slower and slower HB dynamics. This feature means that as d increases, the interface's HB dynamics will converge and eventually tend to bulk water's hydrogen bond dynamics. This behavior is independent of the HB definition as shown by the comparison of the results in panel (a) and (b) of Fig. 5.12.

For comparison, we also calculate the HB dynamics of water molecules at the interface obtained by selecting molecules located at the interface. We call this method Molecule Sampling (MS) (See Appendix E.1 for details). In this method, we first select molecules at the interface at each moment and then make a statistical average of the calculated correlation functions. Specifically, to determine the water molecules staying in the interface layer, we sample at regular intervals, and then calculate the correlation function $C_{\rm HB}(t)$ for these water molecules in the interface layer and then their a statistical average. As the thickness d changes, the $C_{\rm HB}(t)$ for the interface will change. Figure 5.13 shows how the function $C_{\rm HB}(t)$ changes with the thickness d. The panel (a) and (b) use HB definition criterion ADH, and AHD, respectively. Comparing Fig. 5.12 and Fig. 5.13, we found that when we use the method of MS at the water/vapor interface, the dependence of the correlation function $C_{\rm HB}(t)$ on the interface thickness is consistent with that of $C_{\rm HB}^{(s)}(t)$ for large d. Moreover, regardless of the AHD definition or the ADH definition of the HB, this conclusion is basically valid.

Beside the correlation functions $C_{\rm HB}(t)$ or $C_{\rm HB}^{({\rm s})}(t)$ for the interface, we will further examine the correlation functions $C_{\rm HB}(t)$, n(t), k(t) ($C_{\rm HB}^{({\rm s})}(t)$, $n^{({\rm s})}(t)$, $k^{({\rm s})}(t)$), and the rate constants k, k'.

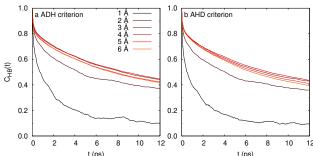


FIGURE 5.13: The $C_{\rm HB}(t)$ for the instantaneous interfacial H-bonds with different thickness d, as computed from (a) the ADH and (b) the AHD criteria. These results are based on MS method. The sampling is performed every 4 ps.

Depth-dependence of k and k' To find the reaction rate constants k and k', we also have two choices. We can sue the correlation functions $C_{\rm HB}^{(\rm s)}(t)$, $n^{(\rm s)}(t)$ and $k^{(\rm s)}(t)$; We can also first select the water molecules at the instantaneous interface at each time point t, and start from the corresponding correlation functions $C_{\rm HB}(t)$, n(t) and k(t) of the H-bonds of these selected water molecules. Figure 5.14 compares the rate constants (k and k') and the lifetime $\tau_{\rm HB}$ obtained by the two different methods, i.e., the instantaneous Interfacial Hydrogen Bond (IHB) and MS methods. We see that, whether it is k, k' or $\tau_{\rm HB}$, their changing trend with the thickness d of the interface is only slightly affected by the calculation methods. To illustrate this point more clearly, we compare the k, k' and $\tau_{\rm HB}$ obtained under the two methods. We listed more detailed data in Table 5.3 to 5.6.

As we can see from Fig. 5.14, when the thickness d is large enough ($d > d_0 \sim 4$ Å), these two constants agree well quantitatively. This result shows that the two extreme

statistical methods (see Appendix E.1) for the HB dynamics of the interface did not produce much difference.

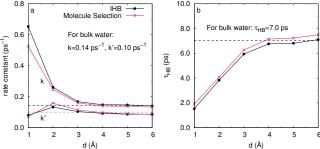


FIGURE 5.14: Dependence of (a) the reaction rate constants k and k' and (b) the HB lifetime $\tau_{\rm HB}$ on the interface thickness d, obtained by the IHB and MS methods, respectively. The corresponding k, k' and $\tau_{\rm HB}$ in bulk water are also drawn with dashed lines as a reference. In panel a, the k of bulk water is represented by a black dashed line, and the k' is represented by a black dashed line; in panel b, the $\tau_{\rm HB}$ of bulk water is represented by a black dashed line. The ADH criterion of H-bonds is used and the least square fits are carried on the time region 0.2 ps < t < 12 ps.

We also found that when we focus on the molecules in the interface layer whose thickness is less than d_0 , the values of the reaction rate constants slightly depend on the method we use. That is, the k obtained by the IHB method is relatively larger than by the MS method, and k' is smaller. Since $\tau_{\rm HB} = 1/k$, large k directly leads to a relatively shorter HB lifetime. This result is related to our definition of IHB, and it is the same as our expectation: The definition of interfacial H-bonds (or $h^{(\rm s)}(t)$) makes the HB break rate on the interface artificially increased. At the same time, the MS method retains the original rate constant of H-bonds, but it may include the contribution of bulk water molecules to the rate constant. That is why the MS method slightly underestimate the k.

In Fig. 5.14, the k, k' and $\tau_{\rm HB}$ for the *bulk* water are also drawn with dashed lines as a reference. Comparing the above-mentioned physical quantities for the water/vapor interface and bulk water, we found that when the interface thickness d is larger then d_0 , no matter which statistical method is used, the value of the calculated reaction rate constants of the interface water is *greater* than that in bulk water. Therefore, since the HB lifetime can be calculated by the relation $\tau_{\rm HB} = 1/k$, the value of $\tau_{\rm HB}$ for water molecules at the water/vapor interface is smaller than that for bulk water.

Furthermore, we have found from Fig. 5.14 that as the interface thickness d increases, the values of k and k' also tend to the values of rates in bulk water at the same condition. These results are obtained by the least squares method in the same interval (0.2–12 ps). This verifies that the IHB method can get as good results as the method of MS for $d > d_0$. Because the method of IHB is easier to operate, it can be used to calculate the HB dynamics and thus HB lifetime for the water/vapor interface when d is larger than d_0 . For the water/vapor interface, d_0 is approximately 4 Å or equals to the size of 2–3 water molecules. This result coincides with the previous theoretical results based on MD simulation[131, 215, 216] and VSFG spectroscopy[217, 218], which have shown that isotropic properties are already present at sub-nanometer distances from the surface of ordered hydrophobic monolayers. They suggested a "healing length" of about 3 Å with the bulk-phase properties of water recovered within the top few monolayers. We also noticed that the method of MS depends on our sampling rate on the trajectory of the simulated system, while the

IHB method does not require such sampling. Therefore, the IHB method is a more convenient method to determine the HBD at the water/vapor interface.

Finally, because the real HB dynamical properties of interface molecules are between the results of the above two methods, we can approximate the interfacial HB dynamics, by either the IHB and the MS method if the thickness of the interface is large enough, i.e., $d > d_0$.

TABLE 5.3: The k and k' for the interfacial HB dynamics of the water/vapor interface, through the IHB method, with ADH criteria. We carried on the longer time region 0.2 ps < t < 12 ps (same below).

d (Å)	$k \text{ (ps}^{-1})$	$k' \; (ps^{-1})$	$\tau_{\rm HB} (=1/k) \; ({\rm ps})$
1.0	0.653	0.080	1.53
2.0	0.261	0.133	3.83
3.0	0.168	0.104	5.94
4.0	0.148	0.092	6.76
5.0	0.147	0.087	6.81
6.0	0.139	0.087	7.17

TABLE 5.4: The k and k' for the interfacial HB dynamics of the water/vapor interface through the IHB method, with AHD criteria.

d (Å)	$k \; (\mathrm{ps}^{-1})$	$k' \text{ (ps}^{-1})$	$\tau_{\rm HB} (=1/k) \; ({\rm ps})$
1.0	0.661	0.080	1.51
2.0	0.265	0.133	3.77
3.0	0.172	0.102	5.82
4.0	0.148	0.090	6.74
5.0	0.149	0.084	6.72
6.0	0.144	0.078	6.93

TABLE 5.5: The k and k' for the interfacial HB dynamics of the water/vapor interface through the MS method, with ADH criteria.

d (Å)	$k \; (ps^{-1})$	$k' \text{ (ps}^{-1})$	$\tau_{\rm HB} (=1/k) \; ({\rm ps})$
1.0	0.526	0.072	1.90
2.0	0.246	0.158	4.07
3.0	0.160	0.114	6.26
4.0	0.140	0.097	7.15
5.0	0.138	0.090	7.24
6.0	0.133	0.085	7.49

TABLE 5.6: The k and k' for the interfacial HB dynamics of the water/vapor interface through the MS method, with AHD criteria.

d (Å)	$k \text{ (ps}^{-1})$	$k' \; (ps^{-1})$	$\tau_{\rm HB} (=1/k) \; ({\rm ps})$
1.0	0.610	0.083	1.64
2.0	0.235	0.142	4.62
3.0	0.138	0.102	7.22
4.0	0.141	0.098	7.07
5.0	0.120	0.078	8.40
6.0	0.119	0.071	8.39

5.4. Summary 51

5.4 Summary

In this chapter, we have introduced the interfacial HB population operator, and correlation functions $C_{\rm HB}(t)$, $S_{\rm HB}(t)$, n(t), and k(t) based on these operators. Then the HB dynamics calculations for the bulk water and the water/vapor interface are based on these correlation functions. The $C_{\rm HB}(t)$ describes the relaxation of H-bonds. The $S_{\rm HB}(t)$ gives the average lifetime $\langle \tau_a \rangle$ of the continuous H-bonds. The n(r) describes the probability that a HB is broken but still separated by cutoff radius $r_{\rm OO}^c$. The k(t) describes the rate that an initially HB breaks at time t. Based on DFTMD simulations, starting from the functions $C_{\rm HB}(t)$, n(t), and k(t), we have calculated the reaction rate constants k and k' for HB rupture and regeneration for bulk water and the water/vapor interface.

For the water/vapor interface, we have studied the HB dynamics for instantaneous interfaces using two different statistical methods, the MS and the newly introduced IHB. We have found that as the thickness of the interface layer increases, the HB reaction rate constants tend to the rate constants in the bulk water. From above results for the water/vapor interface, we conclude that from the perspective of HB dynamics, the thickness of the water/vapor interface is about 3 Å. This value is smaller than that obtained from the SFG spectra (Ref. paragraph 4.1) for the LiNO $_3$ /vapor interface. As the presence of ions in solutions affects the HB network, we will analyze these effects in the next chapter.

Chapter 6

Hydrogen bond dynamics in electrolyte solutions

In this chapter, we explore the effects of nitrate ions, iodide ions and alkali metal cations on the HB dynamics and water reorientation dynamics at the interface of alkali nitrate solutions and alkali iodine solutions. In paragraphs 6.1, we discuss the ion-water HB dynamics for the whole interface. Then in paragraph 6.2, we study the HB dynamics and HB lifetime specifically in the solvation shells of ions. In paragraphs 6.3 and 6.4, we focus on the reorientation dynamics of water molecules at the interfaces and in the solvation shells of ions in the solutions. All simulations in this chapter were performed at the DFT (BLYP + D) level and at 300 K within the canonical NVT ensemble. The length of each trajectory is about 60 ps. For each solution, we have performed simulations for the solution/vapor interface and bulk system, respectively. The detailed parameter settings of different systems can be found in Appendix B.

6.1 Ion-water HB dynamics at electrolyte/vapor interface

6.1.1 Lithium nitrate solutions

We simulated a LiNO₃ solution/vapor interface including Li⁺ and NO₃⁻, as shown in Fig. 6.1(A) (water molecules are not shown). For computational details, see Appendix. B The correlation functions $C_{\rm HB}(t)$ and $S_{\rm HB}(t)$ for both nitrate—water (N–W) and water—water (W–W) hydrogen bonds in the slab are shown in Fig. 6.2 a and b, respectively. For both HB definitions, we found that the decay of $C_{\rm HB}(t)$ and $S_{\rm HB}(t)$ for N–W hydrogen bonds is much faster than that for W–W hydrogen bonds. From the relation 5.7, the faster relaxation of $S_{\rm HB}(t)$ implies that the N–W hydrogen bonds have shorter lifetime than the W–W hydrogen bonds in bulk phase. The calculation results obtained from the simulations are in good agreement with numerous experimental and simulation results[137, 177, 186, 219]. It means that nitrate has structure-breaking ability, or compared with W–W interaction, that the N–W interaction is weaker.

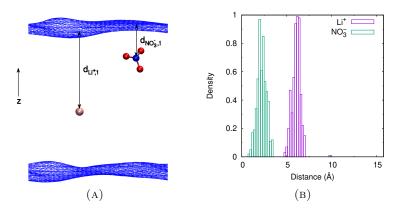


FIGURE 6.1: Distribution of ions at the LiNO₃ solution/vapor interface. (A) Distances between ions and one of the instantaneous surfaces (blue meshes) for a slab of aqueous LiNO₃ solution. (B) Density distribution the Li⁺–surface and NO₃⁻–surface distances at the LiNO₃ solution/vapor interface. The horizontal axis represents the distance between the ion and the instantaneous surface, which is defined in Eq. 6.1. The distance refers specifically to $d_{X,1}$, the distance between the ion X and one of the instantaneous surfaces. Zero distance denotes the instantaneous surface of the interfacial system of LiNO₃ solution.

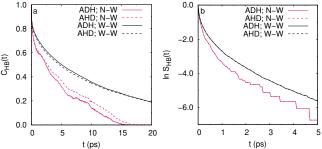


FIGURE 6.2: Time dependence of (a) $C_{\rm HB}(t)$ and (b) $\ln S_{\rm HB}(t)$ of all water—water (W–W) and nitrate—water (N–W) bonds for the slab of LiNO₃ solution, as computed from the ADH (solid line) and AHD (dashed line) criterion of H-bonds. The W–W bonds represents the H-bonds between all pairs of water molecules in the entire slab.

Since the structure-breaking ability of ions is closely related to surface propensity, we speculate that the rate of relaxation of the correlation functions ($C_{\rm HB}(t)$) and $S_{\rm HB}(t)$) of N–W bonds may be related to the distribution of NO $_3^-$ ions at various depths under the surface. Therefore, we calculated the distribution of ions at different depths relative to the instantanous interfaces of the LiNO $_3$ solution.

For the DFTMD trajectory of a solution interface, we can now get instantaneous surface (see blue meshes in Fig. 6.1 (A)). For any molecule or ion X in such a solution/vapor interface, we can define its distance from the instantaneous surface. We assume that the z-axis is the normal direction, then the distances between the particle and the instantaneous surfaces are:

$$d_{X,1}(t) = z_{X,1}^{\text{surf}}(t) - z_X(t), \tag{6.1}$$

$$d_{X,2}(t) = z_X(t) - z_{X,2}^{\text{surf}}(t), \tag{6.2}$$

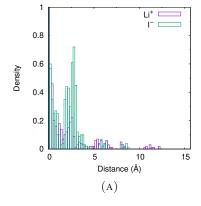
where $z_{\rm X}$ is the coordinate of the particle X in the normal direction at time t, $z_{{\rm X},i}^{\rm surf}(t)$ is the z coordinate of the surface position corresponding to particle X at time t, and the subscripts i=1 and 2 respectively identify the lower and upper instantaneous surfaces. As an example, Fig. 6.1 (A) shows the distance, $d_{{\rm Li}^+,1}$ ($d_{{\rm NO}_3^-,1}$), between ${\rm Li}^+$ (${\rm NO}_3^-$) ions and one of the instantaneous surfaces at a certain moment.

Figure 6.1 (B) shows the probability density of the distance between the ion and the instantaneous surface for the LiNO₃ solution/vapor interface. We found that when the system reaches an equilibrium state, the Li⁺ ion is stable within a few angstroms below the instantaneous interface, while the NO₃⁻ ion resides near the surface. When the system is in equilibrium (t > 10 ps), the distance $d_{NO_3^-,1}$ is around 2 Å, which indicates that the nitrate ion is in the top layer of the instantaneous interface.

The result for the HB dynamics are in agreement with what we have learn in paragraph 5.1 from the vibrational spectroscopy. From the calculation of the VSFG spectrum of the LiNO₃ solution/vapor interface and the VDOS of water molecules in the water cluster NO₃ (H₂O)₃, we have shown that (1) Compared with the water/vapor interface, the VSFG spectrum of the LiNO₃ solution/vapor interface has a blue-shifted HB band (see Fig. 4.3); (2) The vibration frequency of water molecules from the LiNO₃ interface is higher than that at the water/vapor interface (see Fig. 4.4). The two conclusions above indicate that nitrate ions have the surface propensity. Here, the probability distribution of the nitrate ion in the slab shows that the average position of the nitrate ion relative to the solution/vapor interface is 2 Å, which is consistent with the conclusion from the VSFG spectra and VDOS calculations.

6.1.2 Alkaline iodide solutions

We also calculated the distribution of ions at different depths at the instantaneous interface for the alkali iodine solutions. Figure 6.3–6.4 shows the density of the distance between the anion (cation) and the instantaneous surface in the simulated the solution/vapor interface of the LiI, NaI and KI solution, respectively. As an example, the distances between ions and one of the instantaneous surfaces (grey meshes) for a slab of aqueous KI solution are shown in Fig. 6.4 (A). We found that, the I $^-$ ion resides near the surface, while alkali metal ions does not have such a strong tendency toward the surface.



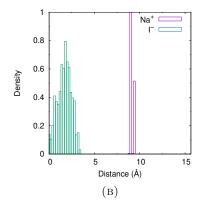


FIGURE 6.3: Distribution of ions at the LiI and the NaI solution/vapor interface. (A) Density distribution the Li⁺-surface and I⁻-surface distances at the LiI solution/vapor interface. (B) Density distribution the Na⁺-surface and I⁻-surface distances at the NaI solution/vapor interface.

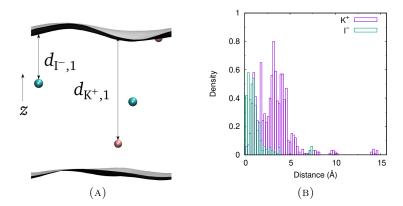


FIGURE 6.4: Distribution of ions at the KI solution/vapor interface. (A) Distances between ions and one of the instantaneous surfaces (grey meshes) for a slab of aqueous KI solution. (B) Density distribution the K⁺-surface and I⁻-surface distances at the KI solution/vapor interface.

The experimental data of the concentration dependence of surface tension $d\gamma/dm_2$ of solutions containing Li⁺, Na⁺ and K⁺ ions also show that they do not have the strong surface propensity of I⁻ ions (see paragraph 2.6). Therefore, they did not show apparent surface propensity on the time scale of our current simulations.

Table 6.1: The average of the continuum HB lifetimes $\langle \tau_{\rm a} \rangle = \int_0^\infty S_{\rm HB}(t) dt$ (unit: ps) in the first hydration shell of I⁻ ion and of alkali metal ion at the interface of three 0.9 M alkali-iodine solutions.

	I^- -shell	cation-shell	bulk water	w/v interface
LiI	0.22	0.24	0.25	0.23
NaI	0.24	0.28	0.25	0.23
KI	0.20	0.23	0.25	0.23

To investigate ions' effects, we calculated the HB population correlation functions at the water/vapor interface and we compared it to that in the ions' slovation shells at aqueous electrolyte interfaces. We consider the $C_{\rm HB}(t)$ and $S_{\rm HB}(t)$, from which the HB relaxation time $\tau_{\rm R}$ and the HB lifetime $\tau_{\rm a}$ can be calculated, respectively. Table 6.1 lists the continuum HB lifetime in the first hydration shell of I⁻ ion and of alkali metal ion at interfaces of alkali-iodine solutions: LiI, NaI and KI. For reference, we also give the $\tau_{\rm a}$ for bulk water and the water/vapor interface in the third and fourth column, respectively. The continuum HB lifetime $\tau_{\rm a}$ in the solvation shell of alkali metal (iodine) ions is larger (smaller) than that of H-bonds at the water/vapor interface. For the LiI solution, the water molecules bound to the cation ion Li⁺, on average, have a continuum HB lifetime $\tau_{\rm HB} \sim 0.24$ ps, which is longer than that of molecules bound to I⁻ or at the water/vapor interface. Besides, we found that $\tau_{\rm a}$ for the I⁻-shell is smaller than that for water/vapor interface in general.

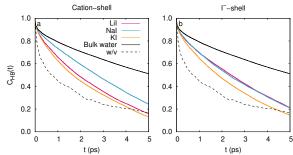


FIGURE 6.5: The $C_{\rm HB}(t)$ of water—water hydrogen bonds in the solvation shell of (a) cations and (b) I⁻ at the interfaces of 0.9 M LiI, NaI and KI solutions, respectively. The solid and dashed line are for the instantaneous 1-Å water/vapor interface (see paragraph 5.3) and bulk water, respectively. The data for bulk water are calculated from water in the middle of the slab of the LiI solution.

Fig. 6.5 a and b show that the $C_{\rm HB}(t)$ of H-bonds within the alkali cations and I⁻ decay faster than those in bulk water. The interface of the LiI solution contains H-bonds between water molecules similar to those in bulk water, i.e., water molecules participating in these H-bonds are not in the solvation shell of ions. This result is consistent with the observation of femtosecond midinfrared pump-probe experiments on the O-H stretch vibration of water molecules in aqueous solution, that changing the nature of the cation does not affect the dynamics of solvating water [151]. It is also in agreement with to the following ab initio simulation results: water molecules that directly surround the cation, the O-H groups point away from the cation and form $O-H\cdots O$ hydrogen bonds with bulk water molecules [151, 220, 221]. From Fig. 6.5 (b), we found that for all three alkali-iodine solutions, the $C_{\rm HB}(t)$ for hydration shell water molecules of I⁻ decays faster than that for molecules in bulk water. The simulation produces similar result as Omta and coworker's experiments of femtosecond pump-probe spectroscopy, which demonstrate that anions (SO_4^{2-} , ClO_4^- , etc) have no influence on the dynamics of bulk water, even at high concentration up to 6 M[222, 223]. Here, we have found that the cations Li⁺ and Na⁺ do not alter the H-bonding network outside the first hydration shell of cations. It is concluded that no long-range structural-changing effects for alkali metal cations.

Besides, the correlation function $C_{\rm HB}(t)$ for the instantaneous interface layer of thickness d=1 Å in pure water is also shown in Fig. 6.5. Its ultrafast relaxation process shows that the effects of ions (alkali cations and I⁻) are not as obvious as that of the water/vapor interface.

Effects of the ion concentration To investigate the effect of the ion concentration we considered one additional system with higer concentration, namely 1.8 M. We calculated the $C_{\rm HB}(t)$ for the interfaces of the alkali-iodine solutions, and the relaxation time $\tau_{\rm R}$ for each of them can be determined according to formula 5.2. Here, the interface means all the water molecules in each model. The $\tau_{\rm R}$ for the interfaces of the LiI (NaI) solutions are given in Table 6.2. Generally, they are in the range 1–10 ps. The values of $\tau_{\rm R}$ decrease as the concentration of the solutions increases.

The concentration dependence of the $S_{\rm HB}(t)$ was also calculated. Figure 6.6 a (6.6 b) gives the $S_{\rm HB}(t)$ for the interfaces of 0.9 M and 1.8 M LiI (NaI) solutions. This result indicates that, for the interface of alkali-iodine solution, the continuum HB lifetime decrease as the concentration of the LiI (or NaI) solution increase. This can be explained as following: As the ion concentration increases, more iodide ions gather on the surface. The ratio of the number of iodide ions to the number of other

Table 6.2: The relaxation time $\tau_{\rm R}$ (unit: ps) of the $C_{\rm HB}(t)$ for the interface of the LiI (NaI) solutions.

concentration	$ au_{ m R}~({ m LiI})$	τ_{R} (NaI)
0	11.50	11.50
$0.9 \mathrm{M}$	7.04	10.60
1.8 M	4.40	1.96

water molecules among the neighbors of the surface water molecules is set to $r_{\rm I}$. For the water molecules in the first solvation shell of the cation, the ratio of the number of cations to other water molecules in its neighbors is set to $r_{\rm cation}$. Water molecules below the surface are surrounded by more water molecule neighbors than molecules at the interface. Since the upper half of the surface is vacuum and the cations here do not aggregate on the surface. Therefore, $r_{\rm I}$ -> $r_{\rm cation}$, so the influence of the cation on the HB relaxation is not as strong as the influence of the I⁻ ion on the surface. We believe that this fact is the reason that the HB relaxation of the water molecules at the interface becomes faster as the alkali iodide concentration increases. Therefore, the HB relaxation of the interface and the lifetime of H-bonds at the interface will be more dominated by the concentration of ions accumulated at the interface.

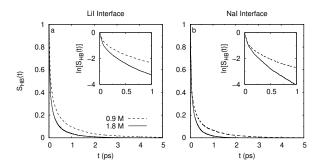


FIGURE 6.6: Time dependence of the $S_{\rm HB}(t)$: (a) LiI solution/vapor interface; (b) NaI solution/vapor interface. The insets show the plots of $\ln S_{\rm HB}(t)$.

To summarize, we have investigated the effect of alkali-nitrate and alkali-iodide on the HB dynamics of water molecules at solution/vapor interfaces, obtained from ab inito simulations. N-W's $C_{\rm HB}(t)$ and $S_{\rm HB}(t)$ decay faster than W-W's, which proves that the N-W hydrogen bonds is weaker than W-W ones. The result we obtained from VSFG spectra in paragraph 5.1 is that the blue shift of the HB band in the ${\rm Im}\chi^{(2)}$, or, the nitrate ion in the solution tends to be distributed on the surface, which is also consistent with the smaller relaxation time given by the correlation functions $C_{\rm HB}(t)$ and $S_{\rm HB}(t)$. Compared with the water/vapor interface, the characteristic relaxation time of H-bonds between water molecules at the interface of the alkali metal salt solution as a whole becomes smaller and increases with the increase of the solute concentration.

6.1.3 Ion-water bond dynamics: breaking and reforming

As for the effect of the interface on the HB dynamics in alkali-iodine solutions, we also calculate the survival probability for interfaces with different sizes of thickness. The result for the LiI solution/vapor interface shows that H-bonds at the water/vapor interface decay faster than that in bulk water. The logarithm of $S_{\rm HB}(t)$ is

given in Fig. E.3 in Appendix E, in which the thickness of the alkali-iodine solutions can be determined. As the interface thickness increases, the $S_{\rm HB}(t)$ converges to a fixed curve, which characterizes the HB dynamics of the solution/vapor interface. In particular, it gives the average continuum HB lifetime in bulk solution.

Alkali nitrate solutions First, let us take a look at the changes in H-bonds of water molecules by nitrate ions in bulk solution. The $\ln S_{\rm HB}(t)$ for the water-water hydrogen bonds and nitrate-water hydrogen bonds at the interface of the LiNO₃ solution is shown in Fig. 6.7. The $\ln S_{\rm HB}(t)$ shows that the nitrate-water bonds are weaker than water-water bonds, i.e., nitrate ions accelerate the HB dynamics in water.

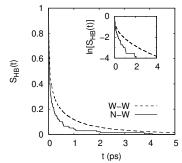


FIGURE 6.7: The $S_{\rm HB}(t)$ of water-water (W-W) and nitrate-water (N-W) bonds at the LiNO₃ solution/vapor interface. The inset is the plot of $\ln S_{\rm HB}(t)$. The results are calculated for the temporal resolution $t_{\rm t}=1$ fs. For the definition of $t_{\rm t}$, see Appendix E.

The difference between nitrate–water and water–water bonds ia also analyzed in terms of the survival probability $S_{\rm HB}(t)[157,\ 192,\ 197]$, reported in Fig. 6.7. The integration of $S_{\rm HB}(t)$ from 0 to $t_{\rm max}=5.0~{\rm ps}[224]$, gives the approximate lifetime $\tau_{\rm a}[225]$. The values of $\tau_{\rm a}$ depend on the temporal resolution t_t , during which the H-bonds that break and reform are treated as intact[210]. Here, we choose the temporal resolution as $t_t=1$ fs. Then, Fig. 6.7 gives $\tau_{\rm a}=0.20~{\rm ps}$ for nitrate–water hydrogen bonds at the interface, and $\tau_{\rm a}=0.42~{\rm ps}$ for water–water hydrogen bonds. This result of $\tau_{\rm a}$ is consistent with the experimental result of Kropman and Bakker ($\tau_{\rm a}=0.5\pm0.2~{\rm ps}$) for water–water hydrogen bonds [226]. The smaller value of $\tau_{\rm a}$ for nitrate–water hydrogen bonds implies that the nitrate–water hydrogen bonds are weaker than bonds between water molecules. This is consistent with the VDOS analysis and the blue-shifted frequency (of 55 cm⁻¹ towards the blue) of the OH stretching in the nitrate-water HB (see Fig. 4.5).

Alkaline iodide solutions Hydrogen bonds between water molecules and other species play decisive role in chemical and biological systems. For this type of HB, some results obtained by molecular simulations have been obtained. For example, the HB dynamics of surfactant—water and water—water hydrogen bonds at the interface has been analyzed by Chanda and Bandyopadhyay[187]. Similar analyse for nitrate—water is also done by Yadav, Choudhary and Chandra by first-principles MD simulations[227]. As we know, in the case of water—water hydrogen bonds, the cutoff radius $r_{OO}^c = 3.5$ Å is the position of the first minimum of the oxygen—oxygen RDF (see Fig. 5.5).

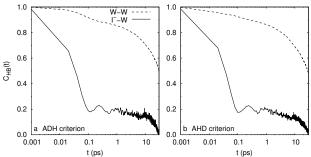


FIGURE 6.8: Time dependence of the intermittent correlation functions $C_{\rm HB}(t)$ of I⁻-water (I⁻-W) and water-water bonds. A base-10 log scale is used for the x-axis.

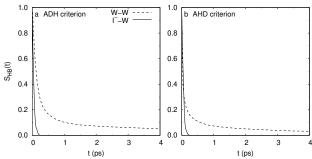


FIGURE 6.9: Time dependence of the continuous correlation functions $S_{\rm HB}(t)$ of I⁻-water (I⁻-W) and water-water bonds.

For the anion-oxygen (X–O) bonds, we can use the similar criteria. The cutoff values for X–oxygen distance are obtained from the positions of the first minimum of the X–oxygen RDF, i.e., $R_{\text{XO}}^{c}=3.7$ and 4.1 Å, for X = nitrate O (see Fig. 6.21) and X = I⁻ (see Fig. 4.8). We have used $\phi^{c}=30^{\circ}$ for the angular cutoff[202]. The function $C_{\text{HB}}(t)$ of I⁻–water and water–water bonds describes the structural relaxation of these H-bonds. The $C_{\text{HB}}(t)$ are shown in Fig. 6.8, and the results of the continuous correlation functions for both definitions (ADH and AHD criterions) for the H-bonds are shown in Fig. 6.9 for I⁻–water bonds. The results of water–water hydrogen bonds are also included for comparison in both Fig. 6.8 and 6.9. For both ADH and AHD definitions of H-bonds, it is found that the I⁻–water bonds show faster dynamics than water–water bonds (consistent to previous MD results by Chowdhuri and Chandra)[202].

The time scales of the relaxation of the I⁻-water bonds are obtained for both definitions. In Table 6.3, we have included the average lifetimes $\langle \tau_a \rangle$ for I⁻-water and nitrate-water bonds. We have performed the fitting in the time region 0.2 ps < t < 12 ps to calculate the forward and backward rate constants for HB reactive flux. No matter from ADH or AHD criteria, the average lifetime $\langle \tau_a \rangle$ of I⁻-water bonds is shorter than that of NO₃-water bonds. In addition, based on the HB population operator for ion-molecule pairs, we also calculated the HB lifetime 1/k for these two hydrogen bonds. The results show that the lifetime of the I⁻-water bonds is only half of lifetime of NO₃-water bonds. Therefore, from the perspective of HB dynamics, we can draw the following conclusion: I⁻-water bonds and NO₃-water bonds are both weaker than water-water bonds. In particular, the I⁻-water bonds is slightly weaker than the NO₃-water ones.

Table 6.3: The dynamical properties of I⁻-water and nitrate-water hydrogen bonds within ADH (AHD) criterion.

Quantities	I^- -water	NO_3^- -water
$\langle \tau_a \rangle$ (ps)	0.10 (0.11)	4.35 (7.91)
1/k (ps)	2.80(2.40)	4.15(6.02)

6.2 Water-water HB dynamics within the ions' solvation shells

We will extend the IHB dynamics (see paragraph 6.3) to H-bonds around ions in aqueous solutions. Similar to the determination of the instantaneous surface, we can define interfaces for molecules or ions in aqueous solutions, i.e., their solvation shells. Below we will combine the interface defined by the second solvation shell of ions, and Luzar-Chandler's HB population [192] to calculate the HB dynamics for the H-bonds between water molecules in the second solvation shell of ions. From characteristics of HB dynamics in the solvation shells, we can obtain the effect of ions on structure and dynamics of aqueous solutions.

Solvation shell HB population Given the solvation shell $\mathbf{k}(t) = \mathbf{k}(\{\mathbf{r}_i(t)\})$, we can define the Solvation shell H-Bonds (SHBs). We use the parameter r_{shell} to denote the radius of the second solvation shell. We define the solvation shell HB population operator $h^{(k,X)}(t) = h^{(k,X)}[r(t)]$ as follows: It has a value 1 when the particular tagged molecular pair are H-bonded and one of the water molecules are inside the solvation shell of species X with a radius r_{shell} , and zero otherwise. The definition of $h^{(k,X)}(t)$ is similar to $h^{(s)}(t)$ in paragraph 5.3 for studying the interfacial H-bonds, and it is used to obtain the dynamic characteristics of H-bonds in the solvation shell of species X with radius r_{shell} .

Similar to the definition of the correlation function $C_{\rm HB}^{(\rm s)}(t)$ in Eq. 5.19, for the H-bonds in an instantaneous interface layer, for a given $r_{\rm shell}$, we define the correlation function $C_{\rm HB}^{(\rm k,X)}(t)$ that describes the fluctuation of the solvation shell H-bonds for ion X:

$$C_{\text{HR}}^{(k,X)}(t) = \langle h^{(k,X)}(0)h^{(k,X)}(t)\rangle/\langle h^{(k,X)}\rangle.$$
 (6.3)

(When not considering specific ions, we denote $C_{\rm HB}^{({\bf k},{\bf X})}(t)$ as $C_{\rm HB}^{({\bf k})}(t)$ for short.) Similarly, we define a correlation function

$$n^{(k,X)}(t) = \langle h^{(k,X)}(0)[1 - h^{(k,X)}(t)]h^{(d,k,X)} \rangle / \langle h^{(k,X)} \rangle,$$
(6.4)

and a reactive flux function

$$k^{(k,X)}(t) = -\frac{dC_{HB}^{(k,X)}}{dt}.$$
 (6.5)

Using these correlation functions, we can determine the rate constants of breaking and reforming and the lifetimes for the solvation shell H-bonding. The results are given in next paragraphs.

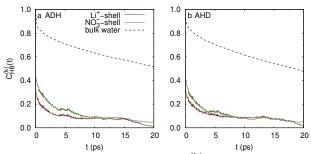


FIGURE 6.10: Correlation function $C_{\rm HB}^{(k)}(t)$ for H-bonds in solvation shells, based on the HB population $h^{(k)}(t)$, as computed from (a) the ADH and (b) the AHD criteria. The $C_{\rm HB}(t)$ (dashed line) in bulk water is also plotted in panel (a) and (b) respectively.

Alkaline nitrate solutions We calculated the HB dynamics for the H-bonds between water molecules in the second solvation shell. The choice of the shell radius comes from the RDFs (see Fig. 6.21). Figure 6.2 a shows that nitrate—water hydrogen bonds, or H-bonds in the first solvation shell of nitrate, is significantly weaker than water-water hydrogen bonds. And from the VDOS for the Li⁺(H₂O)₄ cluster (see Fig. 3.5 (A)), we know that Li⁺-bound water has a red-shifted peak, i.e., the Li⁺water bonds slow down the vibrational relaxations of water-water hydrogen bonds in the solvation shell of Li⁺. If the above two properties were still valid outside the first solvation shell of the nitrate or Li⁺ ions, then we could conclude that the H-bonds in the second solvation shell of nitrate will be weaker than that of the Li⁺ ion, i.e., the relaxation time of $C_{\text{HB}}^{(k,\text{NO}_3^-)}(t)$ will be significantly less than the relaxation time of $C_{\rm HB}^{({\rm k,Li^+})}(t)$. However, as we can see from Fig. 6.10 a and b, the relaxation of $C_{\rm HB}^{({\rm k,NO_3^-})}(t)$ is almost the same as that of $C_{\rm HB}^{({\rm k,Li^+})}(t)$, or the former will not be faster than the latter. Specifically, Fig. 6.10 shows that the relaxation process of H-bonds between the water molecules in the second solvation shell of the nitrate ion is not faster than the relaxation process of H-bonds between the water molecules in the second solvation shell of the Li⁺ion. In general, this result implies that the HB strength between water molecules in the ions' second slovation shells are not affected by the nature of the ions evidently. In other words, the order of the strength of the H-bonds does not significantly affect the surrounding H-bonds.

Alkaline iodide solutions The probability distributions of ions in the solution/vapor interface of the LiI and NaI solutions with repect to the depth of ions in the solutions are calculated. The distributions in Fig. 6.3–6.4 indicate that I⁻ ions prefer to staying at the topmost layer of surface of solutions. The probability distribution shows that I⁻ ions tend to the surface of solutions, while Na⁺ and Li⁺ tend to stay in bulk phase. This result is consistent with the calculations from Ishiyama and Morita[51, 54]. We choose the LiI solution to calculate HB correlation function $C_{\rm HB}^{\rm k}(t)$, and the results are shown in Fig.6.11. Like the LiNO₃ solution, the relaxation functions $C_{\rm HB}^{(\rm k,Li)}(t)$ and $C_{\rm HB}^{(\rm k,I)}(t)$ are very close to each other. This result shows that the presence of ions has no significant effect on the relaxation of H-bonds outside the first solvation shell.

Moreover, the $S_{\rm HB}^{(\rm k,Li)}(t)$ and $S_{\rm HB}^{(\rm k,I)}(t)$ have no significant difference. This result also implies that Li⁺ ions have not significantly affected the relaxation process of H-bonds between the water molecules in the second solvation shells. Comparing the results in Fig. 3.7 in chapter 3, we can summarize the influence of lithium ions on the dynamics of water molecules in the cluster (solution) as follows: Although in water molecule

clusters, the vibration frequency of the water molecules directly connected to the Li⁺ ion has a redshift, the HB relaxation between water molecules in the second solvation shell of the ion is almost no longer significantly affected by the ion.

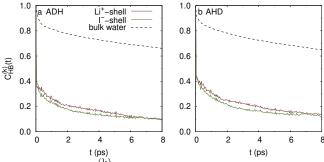


FIGURE 6.11: The $C_{\rm HB}^{(k)}(t)$ for the solvation shell H-bonds with radius $(r_{\rm shell}=5.0~{\rm \AA})$, as computed from (a) the ADH and (b) the AHD criteria. The $C_{\rm HB}(t)$ (dashed line) in bulk water is also plotted in panel (a) and (b) respectively.

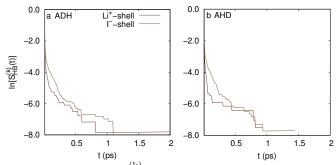


FIGURE 6.12: The $\ln S_{\rm HB}^{(k)}(t)$ for H-bonds in the second solvation shells of Li⁺ and I⁻ ions (in the water/vapor interface) as computed from the (a) ADH and (b) AHD criteria of H-bonds.

6.3 Rotational anisotropy decay of water molecules at the water/vapor interface

The effect of ions on the dynamics of water molecules can also be characterized by rotational anisotropy decay of water molecules. The pump-probe polarization anisotropy monitors electronic alignment [228, 229]. The anisotropy decay can be determined from experimental signal in two different polarization configurations—parallel and perpendicular polarizations, by [230, 231]

$$R(t) = \frac{I_{\parallel}(t) - I_{\perp}(t)}{I_{\parallel}(t) + 2I_{\perp}(t)}$$
(6.6)

where $I_{\parallel}(t)$ is the absorption change when the probe laser pulse is parallel to the pump laser pulse, $I_{\perp}(t)$ is the absorption change when the probe laser pulse is perpendicular to the pump, and t is the time between pump and probe laser pulses. The effects of ion environment on structure and dynamics of water are obtained by comparing the second-order Legendre polynomial, i.e., $P_2(x) = \frac{1}{2}(3x^2 - 1)$, orientational correlation function of the transition dipole. The anisotropy decay can also be obtained by simulations, and calculated by the third-order response functions [232, 233].

If the anisotropy decay is only due to the orientational relaxation of water molecules and within the Condon approximation [234], it is directly related to the orientational correlation function $C_2(t)$ [235–238]

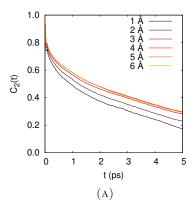
$$R(t) = \frac{2}{5}C_2(t). (6.7)$$

The $C_2(t)$ is given by the rotational time-correlation function

$$C_2(t) = \langle P_2(\hat{u}(0) \cdot \hat{u}(t)) \rangle, \tag{6.8}$$

where $\hat{u}(t)$ is the time dependent unit vector of the transition dipole, and $\langle \rangle$ indicate equilibrium ensemble average [133, 239]. In our simulations, we concentrate on water molecules and consider a unit vector that is directed along the OH bond.

Water/vapor interface For the water/vapor interface, using the method introduced in paragraph 5.3, we first obtained the instantaneous interface with thickness $d = 1, \dots, 6$ Å. Then for each thickness, the method of molecule sampling (see paragraph 5.3) is used to obtain the $C_2(t)$ for the water molecules at the water/vapor interface. The result is shown in Fig. 6.13.



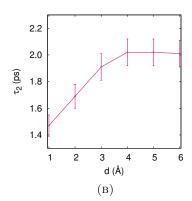


FIGURE 6.13: (A) Time dependence of $C_2(t)$ for water molecules at the water/vapor interface with different thickness in the range of 1 Å to 6 Å. (B) The dependence of decay time τ_2 on the thickness d in the exponential fitting $(t \in [0, 5] \text{ ps})$ of $C_2(t)$ for water molecules at the water/vapor interface.

Here we introduce a fitting with an exponential and provide a decay time τ_2 , i.e.:

$$C_2(t) = Ae^{-t/\tau_2}, (6.9)$$

with A=1.0 for all thickness values $(d=1,\cdots,6\text{ Å})$, and the thickness dependence of τ_2 is shown in Fig. 6.13 (B). We see that when the interface thickness d is small, the decay time τ_2 increases linearly with d. That is, the orientation relaxation process of OH bonds in the thinner interface is faster than in the thicker interface. Similar to the correlations $C_{\text{HB}}(t)$ and $S_{\text{HB}}(t)$ for interfacial water molecules, as the interface thickness increases, the τ_2 converges to a fixed value, which characterizes the decay time of the orientation relaxation process of OH bonds in bulk water. Therefore, we have reached a conclusion that the orientation of OH in the water/vapor interface relaxes faster than in bulk water. From the convergence trend of τ_2 in Fig. 6.13, we

found that at the interface with a thickness greater than 3 Å, the OH orientation relaxation of the water/vapor interface is no longer different from bulk water.

Therefore, starting from the instantaneous interface, we reach a consistent conclusion on the issue of estimating the thickness of the water/vapor interface, no matter from the perspective of OH reorientation relaxation or from HB dynamics.

Lithium nitrate solution The anisotropy decay of OH bonds in water molecules at 0.4 M LiNO3 solution/vapor interface is shown in Fig. 6.14. This result implies that the reorientation relaxation rate of water molecules at the interface of the alkali nitrate solution is very close to that at the water/vapor interface.

We also introduced the fitting with an exponential according to Eq. 6.9. As we known in paragraph 4.1, the LiNO₃ was inserted at one of the two interfaces, we have a model with one salty interface and one neat interface which can be used as a reference. We obtatined $A = 0.70 \pm 0.01$ for both the salty and neat interface, and $\tau_2 = 3.88 \pm 0.09$ ps and 4.04 ± 0.09 ps for the salty and neat water interface, respectively. Therefore, the close reorientation decay time (τ_2) for the salty interface and the water/vapor interface is also obtained from this fitting.

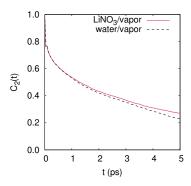


FIGURE 6.14: Anisotropy decay of OH bonds in water molecules at the LiNO₃ solution/vapor interface. The water molecules considered are in instantaneous layer with d=2 Å below the surface.

To understand the reason of the difference between the alkali nitrate solution/vapor interface and the water/vapor interface, we consider the water dangling OH bonds (or free OH bonds) at the interfaces. First, we define a new quantity n, which denotes the number of H-bonds owned by a OH group at a given instantaneous interface. The distribution can be calculated by

$$P(n) = \frac{\sum_{j}^{J} N_{l,j}}{\sum_{i=1}^{J} \sum_{i=1}^{M} N_{i,j}},$$
(6.10)

where $N_{l,j}$ denotes the number of OH bonds which own l HBs at the j-th sampling time, and $l=1,\dots,l_{\max},\ l_{\max}$ is the largest number of HBs that a OH bond can form. For the solution/vapor interface, we use $l_{\max}=5$.

For the LiN solution/vapor interface, the distribution P(n) is shown in Fig. 6.15 a–d.

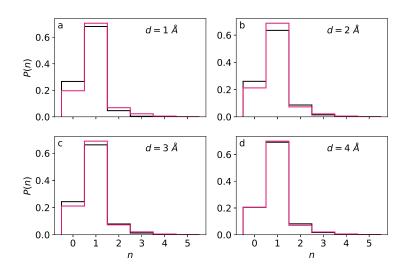


FIGURE 6.15: Distribution P(n) of the number of H-bonds owned by a OH group at instantaneous layers with thickness d ($d = 1, \dots, 4$ Å). The red (black) line is for the LiNO₃ solution/vapor (water/vapor) interface.

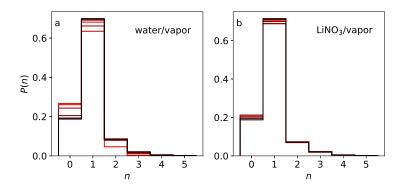


FIGURE 6.16: Distribution P(n) of the number of H-bonds owned by a OH group at instantaneous layers with thickness d ($d = 1, \dots, 6$ Å). The red (black) line is for the interfaces (bulk) phase.

From Fig. 6.15 and 6.16, we found the following features: (1) For thinner layers (Fig. 6.15 a–c, i.e., $d \leq 3$ Å), P(n=0) for LiNO₃ solution/vapor interface is smaller than that for the water/vapor interface. This result implies that there are less water dangling OH bonds at the LiNO₃ solution/vapor interface, compared to the water/vapor interface. (2) The P(n=1) for LiNO₃/vapor interface is larger for d is thin enough. This means that the water at the topmost interface layer are more H-bonded than those at the water/vapor interface. (3) In particular, for d=2 Å, we have P(n=2) for LiNO₃ is larger than P(n=2) for water/vapor interface, which is approximately zero. This character means that for water/vapor interface there are less possibility for a OH bond to form two H-bonds at the topmost instantaneous layer. (4) From the distribution P(n), we again obtain the consistent result, i.e., when d>3 Å, the difference between LiNO₃/vapor interface and water/vapor interface disappears. Both distributions converge to P(n) for bulk water.

These results demonstrate that the structural properties in the instantaneous layer with $d \leq 3$ Å is the main source of interfacial properties, such as HB lifetime, anisotropy decays and SFG spectroscopy.

Alkali-iodine solutions We also calculated $C_2(t)$ for the LiI solution/vapor interface (the whole slab). The results of $C_2(t)$ for these solution/vapor interfaces are shown in Fig. 6.17. It decays significantly faster than $C_2(t)$ at the water/vapor interface, indicating that H-bonds at the LiI solution/vapor interface are orientated more frequently than that of the water/vapor interface. From the single exponent fitting by Eq. 6.9, we obtatined $A = 0.73 \pm 0.02$ (0.77 \pm 0.02) and $\tau_2 = 2.1 \pm 0.1$ ps (2.7 \pm 0.1 ps) for the LiI interface (the water/vapor interface). It shows that the LiI accelerate the dynamics of molecular reorientation of water molecules at the interface.

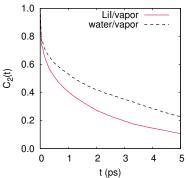


FIGURE 6.17: Time dependence of $C_2(t)$ for OH bonds at the LiI solution/vapor (solid line) and the water/vapor (dashed line) interface. The water molecules considered are in instantaneous layer with d=2 Å below the surface.

From Fig. 6.17, we can find that water molecules at the LiI solution/vapor interface have larger reorientation relaxation rate than that at the water/vapor interface.

The distribution of the number of HBs per OH bond for the LiI solution/vapor interface is also calculated and analyzed. For the LiI solution/vapor interface, the distribution P(n) is shown in Fig. 6.18 a-d.

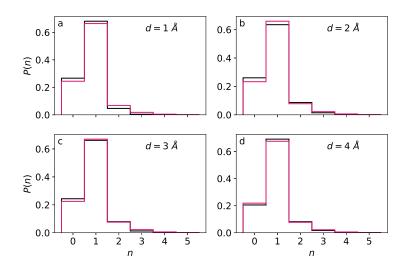


FIGURE 6.18: Distribution P(n) of the number of H-bonds owned by a OH group at instantaneous layers with thickness d ($d = 1, \dots, 4$ Å). The red (black) line is for the LiI solution/vapor (water/vapor) interface.

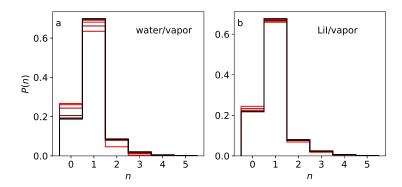


FIGURE 6.19: Distribution P(n) of the number of H-bonds owned by a OH group at instantaneous layers with thickness d ($d = 1, \dots, 6$ Å). The red (black) line is for the interfaces (bulk) phase.

(1)From both Fig. 6.18 and Fig. 6.19, we found that the ratio of water dangling OH bonds for the LiI/vapor interface is smaller than that for the water/vapor interface. (2) We can also find that when d > 3 Å the distribution P(n) does not change anymore. (3) At top layer, P(1) is larger than that for water/vapor interface, this result implies that OH bonds are bonded to ions. From the probability distribution of ions for the LiI solution/vapor interface, see Fig. 7.3, which shows that I^- ions has more propensity for surface than Li^+ cations. We conclude that there are more $O-H\cdots I$ bonds at the LiI solution/vapor interface than in bulk phase. (4) We can find that P(0) of the LiI solution/vapor interface is larger than that of the water/vapor interface, this implies that at the LiI/vapor interface, the $O-H\cdots I$ bonds replace the role of H-bonds at these layers.

Therefore, the increasing density of O-H···I bonds, and decreasing density of

water dangling OH bonds is the two clear structural properties of alkali iodide solution/vapor interface. The decreasing of water dangling OH bonds can explain reduction of the intensity of the $3700~\rm cm^{-1}$ peak in the alkaline iodide solutions, and the increasing density of O–H···I bonds can explain the blue-shift of the 3200– $3600~\rm cm^{-1}$ band.

6.4 Rotational anisotropy decay of water molecules in ions' solvation shells

In this paragraph, we answer the question whether the water molecules in the first solvation hydration shell of different ions have different orientation dynamics.

We calculated the anisotropy decay of water molecules in ions' hydration shells in alkali-nitrate (alkali-iodide) solution/vapor interfaces. The calculated average of $C_2(t)$ is shown in Fig. 6.20. It is obtained by averaging the C(t)'s for 6 different trajectories. The radius of the hydration shell of nitrate O, Li⁺, and water molecules are taken as 4.0, 2.8 and 3.5 Å, respectively. These values are obtained from the position of the first minimum of the RDFs $g_{\rm NN-OW}(r)$ (NN: Nitrate Nitrogen), $g_{\rm Li-OW}(r)$ and $g_{\rm OW-OW}(r)$, for bulk alkali nitrate solution, as shown in Fig. 6.21. To avoid that the considered molecules diffuse out of the first solvation shell, we only consider trajectories with a duration of 10 ps.

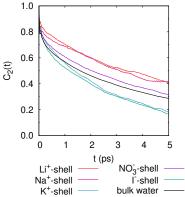


FIGURE 6.20: The $C_2(t)$ for water molecules in the solvation shell of Li⁺, Na⁺, K⁺,I⁻ and nitrate ions at the alkali-nitrate (alkali-iodide) solution/vapor interfaces. For comparison, the $C_2(t)$ for bulk water (black line) is also shown.

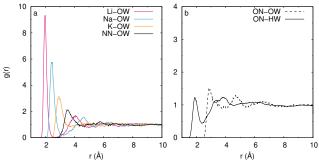


FIGURE 6.21: RDFs for the alkali-nitrate solutions.

For the NaNO₃ solution/vapor interface, the average values of the $C_2(t)$ are also shown in Fig. 6.20. The radius of the hydration shell of nitrate O, Na⁺, and water

molecules are taken as 4.0, 3.2 and 3.5 Å, respectively. (Also in this case ,these values are obtained from the RDFs $g_{\text{NN-OW}}(r)$, $g_{\text{Na-OW}}(r)$ and $g_{\text{OW-OW}}(r)$, see Fig. 6.21.)

Finally we also did the same calculation of $C_2(t)$ for the KNO₃ solution/vapor interface. The average values of $C_2(t)$ are shown in Fig. 6.20. The radius of the hydration shell of nitrate O, K⁺, and water molecules are taken as 4.0, 3.6 and 3.5 Å, respectively. (Also in this case, these values are obtained from the RDFs $g_{\text{NN-OW}}(r)$, $g_{\text{K-OW}}(r)$ and $g_{\text{OW-OW}}(r)$, see Fig. 6.21.)

From the above calculation of $C_2(t)$, we found that in the interfacial systems of alkali metal nitrate solution, nitrate ions always accelerate the reorientation dynamics of water molecules in the hydration shells of nitrate ions. However, the reorientation dynamics of water molecules in the hydration shells of alkali metal ions may slow down (for LiNO₃ and NaNO₃ solutions, respectively) or accelerate (for the KNO₃ solution), due to the presence of alkali metal ions.

For the dynamic process in a short time, we found a relation between the reorientation relaxation time and the radius of hydration shell. Using a single exponential fit of the decay function $C_2(t)$ we obtained the values of τ_2 as reported in Table 6.4. The anisotropy decay is a single exponential given by

$$C_2(t) = e^{-t/\tau_2}.$$

It can be seen that the reorientation relaxation time τ_2 of the water molecules in the hydration shell decreases with the increase of the radius of the hydration shell, as shown in Fig. 6.22. We also give a linear regression function (solid line) to fit this relation, obtained from the data from alkali nitrate (iodide) solutions. The relation between orientation relaxation time τ_2 and the radius is as following:

$$\tau_2(r) = -ar + b,\tag{6.11}$$

where $a = -1.5 \pm 0.5$ and $b = 8.6 \pm 1.7$.

Table 6.4: The radius r of hydration shells and corresponding relaxation times τ_2 at the interface of alkali-nitrate (alkali-iodide) solutions.

ion (molecule)	r (Å)	$\tau_2 \text{ (ps)}$
Li ⁺	2.8	4.57(3)
Na^{+}	3.2	4.35(2)
K^{+}	3.6	2.37(2)
$_{\mathrm{H_2O}}$	3.5	2.92(2)
NO_3^-	4.0	3.32(2)
I-	4.3	2.22(2)

6.5. Summary 71

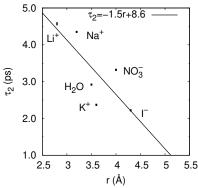


FIGURE 6.22: Dependence of τ_2 on the radius of the solvation shell of molecules and ions (water, Li⁺, Na⁺, K⁺, I⁻ and nitrate ions) in the slab of the alkali nitrate (iodide) solutions.

Using the same approach as above, we calculated τ_2 with a single exponential fitting. We also found that τ_2 of the water molecules in the hydration shell decreases with the increase of the radius of the solvation shell (see the data for I⁻ in Table 6.4).

To expain the $C_2(t)$ for the LiNO₃ and the LiI solution/vapor interfaces, we list the peaks of RDFs for the LiI/vapor, the LiNO₃/vapor and the water/vapor interfaces (see Fig. 6.5). We found that there is a correlaton between reorientation and the strength of the H-bonds: The N-W and W-W hydrogen bonds have very close local structure, while the iodide-water (I⁻-W) HB is of significantly different local structure. This result can be found from the larger value (3.61 Å and 2.66 Å, see the first two rows in Table 6.5) of the first peaks' locations in $g_{\text{I-W}}$, and close first peak locations for ON-OW and OW-OW RDFs (2.80 Åand 2.79 Å, see the 3rd and 5th row in Table 6.5). The difference between water molecules with OW-OW and ON-OW bonds is discussed in paragraph C.3.

TABLE 6.5: The peaks (unit: Å) of RDFs for the LiI/vapor, the LiNO₃/vapor and the water/vapor interfaces (see Table C.5 and C.6).

g(r)	1st peak	2nd peak
I-OW	3.61	-
I-HW	2.66	4.07
ON-OW	2.80	-
ON-HW	1.83	3.30
OW-OW	2.79	-
OW-HW	1.83	3.25

6.5 Summary

We have investigated effects of alkali-nitrate and alkali-iodide on the HB dynamics of water molecules at solution/vapor interfaces. Nitrate—water's HB dynamics decay faster than water—water's, which proves that the nitrate—water hydrogen bonds is weaker than water—water bonds. The result from calculated VSFG spectra is that the blue shift of the HB band in the nonlinear susceptibility, or, the nitrate ion in the solution tends to be distributed on the surface, which is consistent with the smaller relaxation time given by the HB population correlation functions $C_{\text{HB}}(t)$ and $S_{\text{HB}}(t)$.

From a detailed analysis of the dynamics of water molecules in the second solvation shell of ions, we found that the HB relaxation between water molecules in the second solvation shell of the ion is almost no longer significantly affected by the ion. However, the reorientation relaxation time τ_2 of the water molecules in the solvation shell decreases linearly with the increase of the radius of the ions' solvation shell (For the nitrate ion, we use the O–OW distance, instead of the N–OW distance to define the radius of the ion's solvation shell).

The rotational anisotropy decay of water molecules at the water/vapor and solution/vapor interfaces was investigated to understand the effects of ions on the dynamics of water at interfaces. The instantanous interfaces are used in the analyse. The main result is that the ratio of free OH bonds for the water/vapor interface is an important factor that affect the decay rate of the reorientation relaxation of water molecules. Single exponential decay is a good model for water molecules at the interface of the alkali-iodine solutions, and the faster anisotropy decay of water molecules at the interface is the effects of free OH stretch and Hydrogen–Iodide (H–I) bond at the interface. This difference of HB structure from the water/vapor interface is the source of the HB dynamics as well as the $\text{Im}\chi^{(2)}$ spectrum of the interface of alkali-iodine solutions.

Chapter 7

Summary

Using DFTMD simulations, we have analyzed the interfacial structure and dynamics of electrolyte solutions containing alkali nitrates. In particular we have presented a detailed analysis of the VSFG spectra, HB dynamics, and reorientation dynamics of water molecules at the solution/vapor interfaces. We have calculated the interface vibrational spectra in order to provide a molecular interpretation of available experimental data. In view of the similarity between the iodide ion and the nitrate ion in the Hofmeister sequence, we also did the same calculation for the electrolyte solutions of alkaline iodides.

We have shown that the use of simple models, such as small cluster is not suitable to reproduce the experimental spectra and cannot provide a microscopic interpretation of the VSFG spectra. Realistic models of the interface are required to address the perturbation of ions on the water surface. The elucidated mechanism is possibly more general to anions which have high surface propensity for the solution/vapor interface, for example the nitrate ion and the iodide ion.

As a first system we have analyzed the behaviour of a salty interface containing LiNO₃. Both measured and calculated VSFG spectra shows a reduced intensity of the lower frequency portion region, when compared to the water/vapor interface. This reduction is attributed to the H-bonds established between the NO_3^- and the surrounding water molecules at the interface. This effects is only related to the presence of NO_3^- at the water surface and is not affected by the presence of alkali metal ions. Indeed we have shown that although the Li^+ can reside relative close to the water surface, also forming a water mediated ion pair with NO_3^- , its effect on the VSFG spectrum is not visible. The water molecule which mediate the interaction between the NO_3^- and the Li^+ would produce a red-shifted peak in small water cluster, but its influence is not visible in the VSFG spectra. To verify this conclusion, we have calculated the free energy of different configurations for larger water clusters by the Blue-Moon method. The results give consistent results: Li^+ – NO_3^- ion pairs separated by a water molecule have lower free energy than the configuration in which Li^+ is in direct contact with NO_3^- .

For bulk system, based on the Luzar-Chandler HB population operator, we calculated the correlation functions $C_{\rm HB}(t)$, n(t), and k(t), and obtained the rate constants with the HB formation and breakage, and then obtained the information about the HB lifetime. As we did in paragraph 6.1, we studied the population operators of nitrate ion–water hydrogen bond and iodide ion–water hydrogen bond and their correlation functions.

Based on the simulations, the MS method can *partially* gives information on the HB breaking and reforming rates through the water/vapor interface and therefore partially shows how much the interface affects the dynamics of H-bonds in water. This method *underestimates* the HB breaking rate constant of the water/vapor interface. We have provided a method based on the *instantaneous* interface and the

new-defined interfacial HB population operator to obtain the interfacial HB dynamics of instantaneous water/vapor interfaces. Using the correlation functions based on IHB population operator, we directly calculated the HB dynamics of water layer with a certain thickness under the instantaneous surface. The IHB method allows us to avoid choosing which molecules reside in the interfacial layer and it also can provide partial information on the HB breaking and reforming rates at the interface. However, it overestimates the HB breaking rate constant. The calculation results in these two extreme cases indicate that the HB breaking and reforming rate constants in the water layer below the surface tends to be uniform (see Fig. 5.14 a) as the thickness of the water layer increases. The real HB dynamical characteristics at the water/vapor interface are derived from the two extreme cases. In particular, the result gives an estimate of the thickness (\sim 3 Å) of the water/vapor interface.

The combination of the methods of MS and IHB is extended to the solution interface. It has been confirmed that different ions in the electrolyte solution interface have their own unique distributions in the normal direction of the interface. Therefore, the thickness of the solution is greater than the thickness of the water/vapor interface. In addition, we regarded the solvation shell of an ion as an interface, and defined the SHB population operator, which is an extension to the IHB population.

The difference between the HB dynamics for H-bonds in the second solvation shell of the ${\rm Li^+}$ and for nitrate-water H-bonds at the interface is not visible from the values of the HB relaxation time. They reflect the difference between HB dynamics in bulk water and at the water/vapor interface. For the alkaline iodide solution/vapor interfaces, we find that the cations does not alter the H-bonding network outside the first hydration shell. It is concluded that no long-range structural-changing effects for alkali metal cations. Moreover, we use the SHB method to calculate the HB dynamics for water molecules in the second solvation shell. As far as the types of ions (${\rm Li^+}$, ${\rm Na^+}$, ${\rm K^+}$, ${\rm NO_3^-}$ and ${\rm I^-}$ ions) are concerned, the HB dynamics for water molecule pairs does not vary significantly with the type of ions.

From the results of nonlinear susceptibilities, which shows bonded OH-stretching peaks with higher frequencies, we conclude that the water molecules at the interfaces of the alkaline iodide solutions are participating in weaker H-bonds, compared with those at the water/vapor interface. This conclusion is based on the DFTMD simulations, and the origin of the characteristics come from a unique distribution of the iodine ion and the alkali metal cations, which form a double layer [181] over the thickness on the order of 5 Å (see Appendix E). Therefore, both the VSFG spectra of alkaline nitrate and iodide solutions can be explained in the same way, in which the anions has propensity for the surface, and the weaker anion-H bonds at the topmost interface layer contribute to the blue-shifted H-bond band, and the relative less free OH bonds produce a lower free OH stretching band. This conclusion is verified by the distribution of the number of H-bonds owned by a OH group at the instantaneous interface layer. The main difference of this distribution between alkaline nitrate (iodide) solution/vapor interfaces and the water/vapor interface is: the P(n) for $n \neq 3$. This difference can be viewed as the origin of less free OH bonds at these solution/vapor interface, and of weaker H-bonds at the topmost $(d \sim 1-3 \text{ Å})$ interfacial layer. By calculating the rotational anisotropy decay for water molecules at the insantaneous interface, we obtain a main result: the average density of free OH bonds is the most important factor that affect the decay rate of the reorientation relaxation of water molecules.

Appendix A

Calculation of nonlinear optical susceptibilities

Definitions and relations 1. Definition of double product of a m-order tensor A and a n-order tensor B is a tensor with order m + n - 2.

$$A: B = A_{ij}B_{lm}\delta_{il}\delta_{im}. \tag{A.1}$$

2. The components of product of AB is defined by

$$(AB)_{ijlm} = A_{ij}B_{lm}. (A.2)$$

3. For vectors \mathbf{a} , \mathbf{b} , \mathbf{c} and \mathbf{d} , $\mathbf{a}\mathbf{b}$: $\mathbf{c}\mathbf{d} = (\mathbf{a} \cdot \mathbf{d})(\mathbf{b} \cdot \mathbf{c})$. Proof:

$$\mathbf{ab} : \mathbf{cd} = (\mathbf{ab})_{ij} (\mathbf{cd})_{lm} \delta_{jl} \delta_{im}$$

$$= (\mathbf{ab})_{ij} (\mathbf{cd})_{ji}$$

$$= (a_i b_j) (c_j d_i)$$

$$= (\mathbf{b} \cdot \mathbf{c}) (\mathbf{a} \cdot \mathbf{d})$$
(A.3)

Proof of Eq. 2.41 Since

$$\lim_{t \to \infty} |e^{-it(a-ib)}| \le \lim_{t \to \infty} |e^{-bt}| = 0, \tag{A.4}$$

we can obtain

$$\int_0^\infty dt e^{-it(a-ib)} = \frac{1}{b+ia},\tag{A.5}$$

i.e.,

$$i\int_0^\infty dt e^{-it(a-ib)} = \frac{1}{a-ib}. (A.6)$$

Set $a = (\omega_{v'} - \omega_v) - \omega$, and $b = \gamma_{v'v}$, then we have

$$\int_0^\infty dt e^{-it[(\omega_{v'}-\omega_v)-\omega-i\gamma_{v'v}]} = \frac{-i}{(\omega_{v'}-\omega_v)-\omega-i\gamma_{v'v}}.$$

Molecular dipole moment and dipole polarizability derivatives The polarizability tensor α is defined by the relation

$$\delta \boldsymbol{\mu} = \boldsymbol{\alpha} \mathscr{E} \tag{A.7}$$

where $\delta \mu$ is the electric dipole moment (a vector) induced in the molecule by the electric field \mathscr{E} , with the components $\mathscr{E}_x, \mathscr{E}_y$ and \mathscr{E}_z . Now we describe the main algorithm to implement the parametrization of the molecular dipole moment derivative $\frac{\partial \mu_k}{\partial r}$ and dipole polarizability derivative $\frac{\partial \alpha_{\eta\xi}}{\partial r}$. This result can be used in the velocity ACF-based method for calculating the VSFG spectroscopy intensity.

Given a DFT MD trajectory of total length $\sim 10^2$ ps for bulk water, sampled with a frequency $\sim 1 \text{ ps}^{-1}$. For the j-th water molecule in the n-th snapshot of the trajectory, we denote the two OH bonds as $H^{n,j,\epsilon=1}$ and $H^{n,j,\epsilon=-1}$. We will calculate statistical average over all time steps and all OH bonds, therefore, we just denote the corresponding OH bonds by $\epsilon = 1$ and $\epsilon = -1$, respectively. The H atoms in a water molecule are denoted by $H^{\epsilon=1}$ and $H^{\epsilon=-1}$, and the O atom by O^0 .

We used three different frameworks: the lab framework (x^l, y^l, z^l) , the molecular framework (x^m, y^m, z^m) and the bond framework (x^b, y^b, z^b) (see Fig. 2.2). In the lab framework, the z^l -axis is perpendicular to the interface. The molecular frame will be used to decompose the signal into normal modes of water monomers. For the j-th molecule, the z^m axis is along the bisector of the H-O-H angle, the x^m axis is in the molecular plane, and the y^m axis is out of the molecular plane. In the bond framework, $z^{b,\epsilon}$ axis is along the bond ϵ of a molecule, $z^{b,\epsilon}$ is in the molecular plane and $y^{b,\epsilon}$ is out of the molecular plane.

There are two direction cosine matrices between the bond frameworks and molecular framework, we name them as $\mathbf{D}^{\mathrm{b},\epsilon=-1}$ and $\mathbf{D}^{\mathrm{b},\epsilon=1}$ [124], or $\mathbf{D}^{\mathrm{b},-1}$ and $\mathbf{D}^{\mathrm{b},1}$ for short. Then the direction matrix $\mathbf{D}^{\mathrm{b},\epsilon}$ can be represented by direction cosines between $\mathbf{x}^{\mathrm{b},\epsilon}$ and \mathbf{x}^{m} , where $\epsilon=\pm 1$ and θ denotes the H-O-H angle in the j-th water molecule for the n-step:

$$\hat{x}_1^{b,\epsilon} = \epsilon \cos \frac{\theta}{2} \hat{x}_1^m + \sin \frac{\theta}{2} \hat{x}_3^m \tag{A.8a}$$

$$\hat{x}_2^{\mathrm{b},\epsilon} = \epsilon \hat{x}_2^m \tag{A.8b}$$

$$\hat{x}_3^{\mathrm{b},\epsilon} = -\epsilon \sin\frac{\theta}{2}\hat{x}_1^m + \cos\frac{\theta}{2}\hat{x}_3^m \tag{A.8c}$$

i.e.,

$$\mathbf{D}^{\mathrm{b},\epsilon} = \begin{pmatrix} \epsilon \cos\frac{\theta}{2} & 0 & \sin\frac{\theta}{2} \\ 0 & \epsilon & 0 \\ -\epsilon \sin\frac{\theta}{2} & 0 & \cos\frac{\theta}{2} \end{pmatrix}. \tag{A.9}$$

The molecular framework is given by the direction cosine matrix $\mathbf{D}^{m,l}$ (or \mathbf{D}^{m}) between molecular framework and the lab framework

$$\hat{\mathbf{x}}^m = \mathbf{D}^m \hat{\mathbf{x}}^l. \tag{A.10a}$$

The dipole moment of each OH bond with different length is required to determine the dipole moment derivative. Therefore, we elongate (reduce) one bond ϵ by $\Delta r = 0.05 \text{ Å}$ ($\Delta r = -0.05 \text{ Å}$), and keep other bonds in the total system still, then we obtain a updated coordinate. Then the MLWF centers for the system can be calculated from the updated coordinate, using force and energy calculation at the DFT level. From the Wannier centers of the j-th water molecule, we can calculate the dipole moment for the elongated (reduced) bond ϵ . In the bond frame, $|\mu^b| = |\mu_z^b|$. Therefore, we can calculate the k = z component of the dipole moment for ϵ in water molecule j from the MLWF centers [240]. The MLWF centers are computed and the partial dipole

moment for a given molecular species I is defined as [136]

$$\mu^{I} = \sum_{i \in I} (Z_i \mathbf{R}_i - 2 \sum_{n \in i} \mathbf{r}_n^w). \tag{A.11}$$

In particular, here it is expressed as

$$\mu^{b,r+\Delta r,\epsilon} = \frac{1}{2} Z_{\rm O} \mathbf{R}^0 + Z_{\rm H} \mathbf{R}^{\epsilon} - 2r^{w,\epsilon} - 2r^{w,0}, \tag{A.12}$$

where $\epsilon = \pm 1$. In the ϵ frame, the k = z component dipole moment derivative with respect to bond length [241] for the single OH bond ϵ in water molecule j is

$$\frac{\partial \mu^{b,\epsilon}}{\partial r} = (\mu^{b,r+\Delta r,\epsilon} - \mu^{b,r,\epsilon})/\Delta r. \tag{A.13}$$

Since the components of the dipole moment in the molecular framework are given by the transformation:

$$\begin{pmatrix}
\left(\frac{\partial \mu^{m}}{\partial r}\right)_{1} \\
\left(\frac{\partial \mu^{m}}{\partial r}\right)_{2} \\
\left(\frac{\partial \mu^{m}}{\partial r}\right)_{3}
\end{pmatrix} = \mathbf{D}^{b,\epsilon} \begin{pmatrix} 0 \\ 0 \\ \frac{\partial \mu^{b,\epsilon}}{\partial r} \end{pmatrix},$$
(A.14a)

then, to calculate the individual polarizability for a OH bond from Wannier centers, calculations involving finite electric fields (of 0.0001 au intensity) were performed independently along x, y, and z directions at each sampled time step[134]. For the electric field $\mathscr{E} \in \{\mathscr{E}_x, \mathscr{E}_y, \mathscr{E}_z\}$, like in the case of no external electric field, the MLWF centers are calculated. For a finite Δr , the dipole moment is given by

$$\mu^{b,r+\Delta r,\epsilon,\mathscr{E}} = Z_H \mathbf{R}^{\epsilon,\mathscr{E}} + \frac{1}{2} Z^0 \mathbf{R}^{0,\mathscr{E}} - 2\mathbf{r}^{w,\epsilon,\mathscr{E}} - 2\mathbf{r}^{w,0,\mathscr{E}}. \tag{A.15}$$

From the relation (obtained from Eq. A.7)

$$0 = \alpha_{11}^{b,r+\Delta r} \mathscr{E}_1 + \alpha_{12}^{b,r+\Delta r} \mathscr{E}_2 + \alpha_{13}^{b,r+\Delta r} \mathscr{E}_3$$

$$0 = \alpha_{21}^{b,r+\Delta r} \mathscr{E}_1 + \alpha_{22}^{b,r+\Delta r} \mathscr{E}_2 + \alpha_{23}^{b,r+\Delta r} \mathscr{E}_3$$

$$\delta \mu_3^{b,r+\Delta r} = \alpha_{31}^{b,r+\Delta r} \mathscr{E}_1 + \alpha_{32}^{b,r+\Delta r} \mathscr{E}_2 + \alpha_{33}^{b,r+\Delta r} \mathscr{E}_3.$$
(A.16b)
$$(A.16c)$$

$$0 = \alpha_{21}^{b,r+\Delta r} \mathscr{E}_1 + \alpha_{22}^{b,r+\Delta r} \mathscr{E}_2 + \alpha_{23}^{b,r+\Delta r} \mathscr{E}_3 \tag{A.16b}$$

$$\delta\mu_3^{b,r+\Delta r} = \alpha_{31}^{b,r+\Delta r} \mathscr{E}_1 + \alpha_{32}^{b,r+\Delta r} \mathscr{E}_2 + \alpha_{33}^{b,r+\Delta r} \mathscr{E}_3. \tag{A.16c}$$

where

$$\delta \mu_3^{b,r+\Delta r} = \mu_3^{b,\epsilon,r+\Delta r,\mathscr{E}} - \mu_3^{b,\epsilon,r+\Delta r}. \tag{A.17}$$

and the expressions of the electric filed \mathscr{E}^{b} in a OH framework for the three cases of external electric field which is along x, y and z axis in the lab framework, respectively, we obtain 9 equations. For

$$\mathcal{E}^l = (\mathcal{E}_0, 0, 0)^T$$

where $\mathcal{E}_0 = 0.0001$ au, we can obtain the intensity of the external electric filed in the molecular framework

$$\mathscr{E}_1^{\mathrm{m}} = \mathbf{D}_{12}^{\mathrm{m}} \mathscr{E}_2^{\mathrm{l}} \tag{A.18a}$$

$$\mathscr{E}_2^{\mathrm{m}} = \mathbf{D}_{22}^{\mathrm{m}} \mathscr{E}_2^{\mathrm{l}} \tag{A.18b}$$

$$\mathscr{E}_3^{\mathrm{m}} = \mathbf{D}_{32}^{\mathrm{m}} \mathscr{E}_2^{\mathrm{l}},\tag{A.18c}$$

where $\mathbf{D}_{pq}^{\mathrm{m}}$ is the pq-component of \mathbf{D}^{m} . In OH bond framework,

$$\mathscr{E}^{\mathbf{b}} = \mathbf{D}^{\mathbf{b}} \mathscr{E}^{\mathbf{m}}.\tag{A.19}$$

Similarly, we obtain similar (but different) expansions of the intensity of the electric field for the other two cases: when $\mathscr{E}^l = (0, \mathscr{E}_0, 0)^T$ and when $\mathscr{E}^l = (0, 0, \mathscr{E}_0)^T$, respectively. Here, \mathscr{E}_x is the electric field along x-axis in the lab frame. Then the dipole polarizability for the bond ϵ is as follows:

$$\frac{\partial \alpha_{31}^{b,\epsilon}}{\partial r} = (\alpha_{31}^{b,\epsilon,r+\Delta r} - \alpha_{31}^{b,\epsilon,r})/\Delta r \tag{A.20a}$$

$$\frac{\partial \alpha_{32}^{b,\epsilon}}{\partial r} = (\alpha_{32}^{b,\epsilon,r+\Delta r} - \alpha_{32}^{b,\epsilon,r})/\Delta r \tag{A.20b}$$

$$\frac{\partial \alpha_{33}^{b,\epsilon}}{\partial r} = (\alpha_{33}^{b,\epsilon,r+\Delta r} - \alpha_{33}^{b,\epsilon,r})/\Delta r. \tag{A.20c}$$

Therefore, the average for $(\frac{\partial \mu^{\mathrm{b},\epsilon}}{\partial r})_{\kappa}$ and $(\frac{\partial \alpha^{\mathrm{b},\epsilon}}{\partial r})_{\eta\xi}$ (The subscripts $\kappa, \eta, \xi = x^m, y^m, z^m$, or 1,2,3) over all OH bonds gives the molecular dipole and polarizability derivatives.

Appendix B

Computational details of the DFTMD simulations

The DFT calculations were done with the CP2K program[108, 109], which incorporates the Gaussian and Plane Waves (GPW) method[242]. Core electrons were described by Goedecker-Teter-Hutter pseudopotentials[243], while the valence electrons were expanded as a double-zeta Gaussian basis set. The double zete basis sets optimized for the dondesnsed phase[244] were used in conjunction with GTH pseudopotentials[245] and a 280 Ry cutoff for the auxiliary plane wave basis. The discretized integration time step Δt was set to 0.5 fs. The Brillouin zone was sampled at the Γ -point only and, the BLYP XC functional has been employed. All simulations for the water/vapor interface and aqueous solutions were performed within the canonical NVT ensemble.

Bulk water To simulate the bulk water for testing the algorithm of IHB, we constructed the following model. The number of water molecules in the simulation system is 128, the temperature is still T=300 K, and the box is a cube with a side length of 15.64 Å. In this simulation, we relax the value of the target accuracy for the SCF convergence to 10^{-6} . The size of the simulation box is obtained as follows: According to the experimentally obtained relations between water density, temperature and ion concentration (if ions are present), a nonlinear equation set is established, and then the equation set is solved to find the box size.

Lithium nitrate solutions The simulated interfacial system consisted of 127 water molecules and a Li⁺-nitrate pair in a periodic box of size $15.78 \times 15.78 \times 31.56$ Å³, which corresponds to a density of 0.997 g/cm³. At each DFTMD step the corrector was applied only once, which implies just one preconditioned gradient calculation. For a given molecular configuration, $\{\mathbf{r}_i(t)\}$, Eq. 5.17 can be solved through interpolation on a spatial grid.[74] We have taken $\{\mathbf{r}_i(t)\}$ to refer to the positions of all atoms except hydrogen atoms in the system, and because the bulk correlation length of liquid water is about one molecular diameter, we have used $\xi = 2.4$ Å; further, we have used $\rho_0 = 0.016$ Å⁻³, which is approximately one-half the bulk density of water.

This calculation of the $\chi^{(2,R)}$ is done for a model for the water/vapor interface where a slab of 117 water molecules containing one Li⁺ and one NO₃⁻ is included in a period simulation box of 15.60 Å ×15.60 Å ×31.00 Å at 300 K.

For the interface of the LiNO₃ solution, we performed the simulation on a system containing one Li⁺ ion, one nitrate ion and 127 water molecules (concentration: \sim 0.4 M). The size of periodic box is $15.78 \times 15.78 \times 31.56$ Å³.

For the interface of the NaNO₃ solution, the size of periodic box is $15.65 \times 15.65 \times 31.31$ Å³. For the interface of the KNO₃ solution, the size of periodic box is $15.72 \times 15.72 \times 31.43$ Å³.

The simulated bulk alkali metal nitrate LiNO₃, NaNO₃, and KNO₃ solution consisted of one alkali metal ion, one nitrate ion and 127 water molecules in a periodic cubic box of length L, which corresponds to their density of $\sim 1.00~{\rm g~cm^{-3}}$. Here, $L=15.78,\ 15.65,\ {\rm and}\ 15.72$ for LiNO₃, NaNO₃, and KNO₃ solution, respectively.

Water/vapor interface For the calculation of the nonlinear susceptibility $\chi^{(2,R)}$ of water/vapor interface, the water/vapor interface is modeled with a slab made of 121 water molecules in a simulation box of size $15.60 \times 15.60 \times 31.00 \text{ Å}^3$. For this water/vapor interface, the DFTMD simulations at 300 and 330 K are also calculated.

Interface of alkali-iodine solution The calculation of the $\chi^{(2,R)}$ is done for a model for the water/vapor interface where a slab of 118 water molecules containing one Li⁺ and one I⁻ is included in a period simulation box of 15.60 Å × 15.60 Å × 31.00 Å.

Appendix C

Structural characterization of water clusters and solutions

C.1 Water clusters

The structural parameters of the considered water clusters are shown here. Table C.5 gives the average HB lengths r_a (with standard deviations)in $[NO_3 \cdot (H_2O)_3]^-$. Table C.2 (C.3) reports the selected distances characterizing $[NO_3 \cdot (H_2O)_3]^-$ (RNO₃(H₂O)₃), and Table C.4 the selected parameters for RNO₃ (H₂O)₃ (R=Li, Na, K). The unit for length and angle are Å and degree (°), respectively.

Table C.1: Hydrogen bond lengths r_a in $[NO_3 \cdot (H_2O)_3]^-$.

HB bound to	r_a
w1	$2.40\pm0.52;\ 3.02\pm0.72$
w2	$2.56\pm0.48;\ 3.20\pm0.41$
w3	2.29 ± 0.47 ; 3.11 ± 0.72

Table C.2: Parameters of water molecules and H-bonds in $[NO_3\cdot(H_2O)_3]^-$.

water	R_{OH}	∠HOH	$r_{ m OH}$
w1	0.98 ± 0.02	101±4	$2.40\pm0.52,\ 3.02\pm0.72$
w2	$0.98 {\pm} 0.02$	101 ± 5	$2.56\pm0.48,\ 3.20\pm0.41$
w3	0.98 ± 0.02	101 ± 4	$2.29\pm0.47,\ 3.11\pm0.72$

Table C.3: Structural parameters of $\rm RNO_3(H_2O)_3$ from geometry optimization.

Parameters	$LiNO_3(H_2O)_3$	$NaNO_3(H_2O)_3$	$KNO_3(H_2O)_3$
$r_{ m HB1}$	1.67	1.71	1.82
$r_{ m HB2}$	1.91	1.78	1.92
$r_{ m HB3}$	1.82	1.69	1.94
$r_{\text{R-O(w1)}}$	1.91	2.31	2.70
$r_{ ext{R-O(w2)}}$	1.90	2.26	2.70
$r_{\text{R-O(NO}_3^-)}$	1.84	2.29	2.69
∠HOH(w1)	109	106	107
∠HOH(w2)	106	105	105
$\angle HOH(w3)$	108	107	106

TABLE C.5: Lengths of H-bonds in $[NO_3 \cdot (H_2O)_3]^-$. Indices of H atoms: H6, H7 in w1; H9, H10 in w2 and H12, H13 in w3.

HBs	$r_a \pm \delta \ (100 \text{ K})(\text{Å})$	$r_a \pm \delta \ (300 \text{ K}) \ (\text{Å})$
H6-O2	2.75 ± 0.62	2.40 ± 0.52
H7-O4	2.79 ± 0.58	3.02 ± 0.72
H9-O3	2.89 ± 0.60	$2.56 {\pm} 0.48$
H10-O4	2.74 ± 0.49	3.20 ± 0.41
H12-O3	$2.46 {\pm} 0.45$	$2.29 {\pm} 0.47$
H13-O2	2.75 ± 0.59	3.11 ± 0.72

TABLE C.4: Parameters of RNO₃($\rm H_2O$)₃ at 300 K, obtained from the averaging during a DFTMD trajectory. For RNO₃($\rm H_2O$)₃, $R_{\rm OH}$ and $R'_{\rm OH}$ denote the lengths of O-H bonds in which H atoms is H-bonded and is free, respectively.

Parameters	$LiNO_3(H_2O)_3$	$NaNO_3(H_2O)_3$	$KNO_3(H_2O)_3$
$r_{ m HB1}$	1.83 ± 0.14	1.78 ± 0.09	1.82 ± 0.13
$r_{ m HB2}$	2.00 ± 0.25	1.91 ± 0.24	1.80 ± 0.12
$r_{ m HB3}$	1.79 ± 0.16	1.76 ± 0.11	1.89 ± 0.18
$R_{\mathrm{OH}}(\mathrm{w1})$	0.97 ± 0.01	0.98 ± 0.04	0.97 ± 0.03
$R'_{\mathrm{OH}}(\mathrm{w1})$	1.00 ± 0.02	1.00 ± 0.02	1.00 ± 0.03
$R_{\mathrm{OH}}(\mathrm{w2})$	0.97 ± 0.01	0.98 ± 0.02	0.97 ± 0.02
$R'_{\mathrm{OH}}(\mathrm{w2})$	0.99 ± 0.01	1.00 ± 0.02	1.00 ± 0.03
$R_{\mathrm{OH}}(\mathrm{w}3)$	0.97 ± 0.01	0.97 ± 0.02	0.97 ± 0.03
$R'_{\mathrm{OH}}(\mathrm{w3})$	1.00 ± 0.02	1.00 ± 0.02	1.00 ± 0.03
$r_{ ext{R-O(w1)}}$	1.95 ± 0.09	2.34 ± 0.08	2.76 ± 0.11
$r_{\text{R-O(w3)}}$	1.92 ± 0.07	2.32 ± 0.11	2.74 ± 0.13
$r_{\text{R-O(NO}_3^-)}$	1.91 ± 0.08	2.31 ± 0.09	2.74 ± 0.12
∠HOH (w1)	107 ± 4	106 ± 4	105 ± 5
$\angle HOH (w2)$	106 ± 6	105 ± 4	106 ± 4
$\angle HOH (w3)$	108 ± 5	106 ± 3	106 ± 3

Structural and vibrational properties of $[NO_3 \cdot (H_2O)_3]^-$ To find the possible source of the different vibrational features of water molecules in the cluster $[NO_3 \cdot (H_2O)_3]^-$, we consider structural properties and VDOS for water molecules in it. Lengths of H-bonds are shown in Table C.5. The nitrate O (ON)-water O (OW) and nitrate O-water H (HW) RDFs are shown in Fig. C.1.

C.1. Water clusters 83

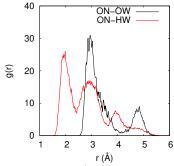


FIGURE C.1: The nitrate O (ON)-water O (OW) and nitrate O-water H (HW) RDFs for $[NO_3 \cdot (H_2O)_3]^-$. Peaks for the former are 1.93, 2.95 and 3.95 Å, and for the later are 2.95 and 4.80 Å.

When T=300 K, the difference r_a between different hydrogen atoms in one water molecule is $\Delta r_a=0.69$ Å, while $\Delta r_a=0.13$ Å for T=100 K. It shows that the vibrational peaks for the three water molecules are much closer than that at the higher temperature 300 K.

The calculated VDOS for water molecules in the cluster at a lower temperature 100 K is given in Fig. C.2. This figure shows that the vibrational peaks for the three water molecules are very close to each other ($\Delta\nu < 10~{\rm cm}^{-1}$) for both vibrational and bending modes. At the lower temperature, the three water molecules are more symmetric distributed bound to the central nitrate. Therefore, the difference between H-bonds in the symmetric isomer of $[{\rm NO}_3\cdot({\rm H}_2{\rm O})_3]^-$ is likely a finite temperature effect, which can be verified by the calculation of the VDOS for water molecules.

Both differences $\Delta \nu$ and Δd decrease as the temperature decrease, Therefore, the different vibrational features are temperature-dependent effect.

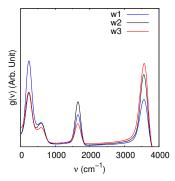


FIGURE C.2: The VDOS $g(\nu)$ for water molecules in the cluster $[NO_3\cdot(H_2O)_3]^-$ at 100 K.

In addition, the VDOS for H atoms and water molecules in $[NO_3\cdot(H_2O)_3]^-$ (Fig. C.3) shows that H's contribution dominates that of the water molecule.

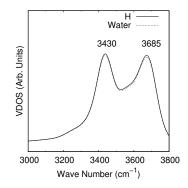


FIGURE C.3: The comparison between the VDOS for H and for a whole water molecule (w1, Fig. 3.1 a), in $[NO_3 \cdot (H_2O)_3]^-$ at 300 K.

C.2 Soultion/vapor interfaces

Dipole orientation distribution of water at aqueous/vapor interfaces Set θ as the angle between the dipole moment of water molecules and the normal vector of the interface, and $P(\theta)$ as the probability. The data used to statistics is the dipole tilt angle θ_i . We use the θ_i instead of $\langle \theta \rangle$ to do the statistics. With picking up θ_l , l=0,10,20,... from the series θ_i , we find that pure water's dipole moment tilt angle is smaller than that of water molecules at the salty water surface. This result means that on the pure water surface, water molecules have more p-polarization components than those in the salty water surface. However, the water molecules at the salty water surface has more s-polarization component. The result is shown in Fig. C.4.

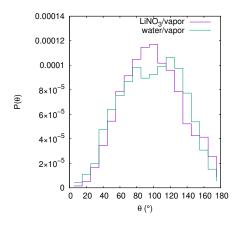


FIGURE C.4: The dipole orientation distribution of water molecules at the water/vapor and the ${\rm LiNO_3}$ solution/vapor interfaces.

RDFs in alkali nitrate solutions RDFs $g_{\text{Li-OW}}(r)$, $g_{\text{Li-HW}}(r)$, $g_{\text{ON-OW}}(r)$ and $g_{\text{ON-HW}}(r)$ in bulk LiNO₃ solution are shown in Fig. C.5.

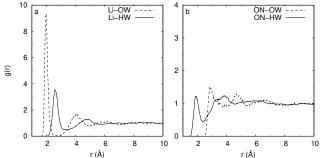


FIGURE C.5: RDFs $g_{\text{Li-OW}}(r)$, $g_{\text{Li-HW}}(r)$, $g_{\text{ON-OW}}(r)$ and $g_{\text{ON-HW}}(r)$ in bulk LiNO₃ solution.

RDFs in alkali iodide solutions RDFs $g_{\text{Li-OW}}(r)$, $g_{\text{Li-HW}}(r)$, $g_{\text{I-OW}}(r)$ and $g_{\text{I-HW}}(r)$ for the interface of the LiI solution are shown in Fig. C.6.

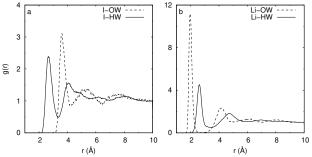


FIGURE C.6: RDFs $g_{\text{Li-OW}}(r)$, $g_{\text{Li-HW}}(r)$, $g_{\text{I-OW}}(r)$ and $g_{\text{I-HW}}(r)$ for the interface of LiI solution.

The radii of second solvation shells are: 5.0 Å for Li⁺, 5.38 Å for Na⁺, and 6.0 Å for I⁻ ions, which are obtained from the RDFs. RDFs g_{X-O} (X=Li⁺, Na⁺, K⁺) for the interfaces of LiI (NaI,KI) solutions are shown in Fig. C.7 a, and the coordination numbers are in Fig. C.7 b. We see that the radius of the solvation shells of Li⁺ ions, Na⁺ ions, and K⁺ ions increase sequentially, and the number of coordination molecules also increase sequentially. However, this order is not true for the relaxation time of HB dynamics between water molecules in the first solvation shell of the ion and other water molecules. As can be seen in Fig. 6.5, the effects of the alkali metal ions and iodide ions are similar, that is, the relaxation time of the HB dynamics in the outer layer is smaller than that in bulk water.

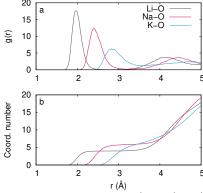


FIGURE C.7: (a) RDFs $g_{X-O}(r)(X=Li^+, Na^+, K^+)$ and (b) the coordination number of Li⁺ (Na⁺, K⁺) ions at the interfaces of LiI (NaI, KI) solution. The coordination number $n_{Li^+}=4$, $n_{Na^+}=5$ and $n_{K^+}=6$.

name	$r_{\rm shell}$ (Å)	coordination number
$n_{ m Li-O}$	3.0	4.0
$n_{ ext{I-H}}$	3.3	5.5
$n_{ m Na-O}$	3.5	6.0
$n_{\text{I-O}}$	4.3	5.8

Table C.6: The coordination number for ions in LiI (NaI) solution.

C.3 Classification of water molecules based on H-bonds

Here we discuss the relation between the reorientation relaxation time of water molecules and their environment. The basic method is as follows: classifying the water molecules at solution/vapor interfaces based on the types of H-bonds and the number of H-bonds.

Alkali nitrate solutions For alkali nitrate solution/vapor interfaces, we can classify the water molecules into three types: nitrate-bound water, water at the water/vapor interface, and bulk water. Because nitrate ions have more surface propensity (see Fig. 4.2), and we have studied the effect of alkali cations on the reorientation relaxation of water molecules, we only consider LiNO₃ solution/vapor interface. We have known that nitrate-bound water are located at the soultion/vapor interface (the thickness ~ 2 Å, see Fig. 4.4). Therefore, among the three types, the first two types of water molecules are interfacial ones. For each type of water molecules, we have chosen six water molecules for obtaining the correlation function $C_2(t)$. The MS sampling method (see paragraph E.1) is used to choose the water molecules for each type.

Lithium iodide solution Following the definition used in Ref.[8], we use the following labels to denote water molecules in an alkali iodide solution: D denotes that the water molecule donates a HB, D' donates a H-I bond, and A accepts a HB. [8] DDAA represents a water molecule with two H-Bonds donated to water molecules and two H-bonds accepted from water molecules (see Fig. C.8 a); DD'AA represents a water molecule with two H-bonds donated to a water molecule and I⁻ (see Fig. C.8 b), and with two H-bonds accepted from other water molecules (see Fig. C.8 c), D'AA represents a water molecule bonded to I⁻ at the water/vapor interface and other H-bonds to water molecules (see Fig. C.8 d). Clearly, we found that D'AA molecules are of free OH stretching during the dynamics.

It is evident that $C_2(t)$ for DDAA and DD'AA molecules do not decay exponentially (Fig. C.9 and Table C.7). This result is similar to the reactive flux function k(t), i.e., the escaping rate kinetics of H-bonds in bulk water. [182] The relaxation of H-bonds in water appears complicated, with no simple characterization in terms of a few relaxation rate constants. Most of the authors believe that the cooperation between neighbouring H-bonds, [196, 246] or self evident coupling between translational diffusion and HB dynamics is the source of the complexity. [182] However, for D'AA molecules at the LiI solution/vapor interface, the $C_2(t)$ decays exponentially,

$$C_2(t) = Ce^{-t/\tau_2},$$

where the amplitude is C = 0.76, and the reorientation rate is $1/\tau_2 = 7.14 \text{ ps}^{-1}$. The single exponential decay of $C_2(t)$ for D'AA molecules, indicates that each D'AA molecule reorientate independently to each other.

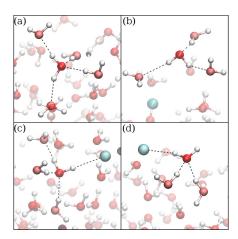


FIGURE C.8: Four types of water molecules at the LiI solution/vapor interface, regarding the HB environments: (a) DDAA; (b) DDA; (c) DD'AA; (d) D'AA. The cyan balls denote I^- ions.

Furthermore, the $C_2(t)$ for D'AA molecules decays much faster than that for DDAA or DD'AA molecules. From the definitions, the D'AA molecule accepts two H-bonds and donates only one H-bond, while both DDAA and DD'AA water molecules own four H-bonds. Therefore, the correlation between H-bonds around the D'AA molecule is weaker than those around the DDAA or DD'AA molecule. Faster decay of $C_2(t)$ for D'AA molecules shows that the reorientation process of D'AA molecules is faster than water molecules in bulk phase, e.g., DDAA or DD'AA molecules.

Additionally, a D'AA molecule has a free OH chemical bond, which can stretch and vibrate freely. This feature is not available in other types of molecules such as DDAA, DD'A, DD'AA, etc. Therefore, at the LiI solution/vapor interface, the most closely related feature of the molecular orientation relaxation process is the number of H-bonds donated by water molecules. The result that D'AA molecules have shortest relaxation time among the four types of water molecules implies that the factor $\langle n \rangle_{\rm HB}$, the average number of H-bonds per water molecule, palys dominate role. This conclusion is consistent with the one obtained from above discussion for LiNO₃/vapor inteface.

Besides, the type (or strength) of HB (water–water, or ion–water) also affects the orientation relaxation process, which is also consistent with the above conclusion about $\text{LiNO}_3/\text{vapor}$ interface. However, from our calculation, the number of H-bonds accepted by water molecules has no major effect on the orientation relaxation of the water molecules at the interface.

Finally, D'AA molecules' inertial-librational motion can not be seen in Fig. C.9. This result implies that the rotational anisotropy decay of D'AA molecules are of the same time scale of the inertial libration, i.e., ~ 0.2 ps. This conclusion can be verified from the value of τ_2 in Table C.7: $\tau_2 = 0.97$ for D'AA molecules, which is smaller than the τ_2 for DDAA, DD'A, and DD'AA molecules.

To summarize, rotational anisotropy decay of water molecules under different local environments is calculated at the LiI solution/vapor interface. The result comes from a different HB types from the usual DDAA HB type in bulk water. The faster anisotropy decay for D'AA molecules reflects the less correlation between different H-bonds for D'AA molecules, which comes from Hydrogen–Iodide bond at the interfaces and the existence of free OH stretching. As we already known from Fig. 4.7, in the LiI solution, I⁻ ions prefer to locate at the interface. Therefore, we infer that the reduction of the inter-correlations between H-bonds occurs at the interface.

Slower rotational anisotropy decay exists for water molecules at the alkali-iodide solution/vapor interfaces, which is the result of a different H-Bond types (D'AA) from DDAA-type molecules in bulk water. The slowing down of anisotropy decay is the effect of Hydrogen-Iodide bonds at the interface. Since the iodide's surface propensity is high, this difference of HB structure from the water/vapor interface changed the $\text{Im}\chi^{(2)}$ spectrum and the total HB dynamics of the interface of alkali-iodide solutions.

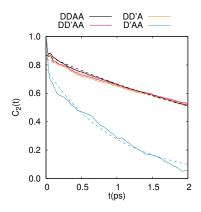


FIGURE C.9: Time dependence of $C_2(t)$ for DD'A, DD'AA, and D'AA water molecules at the LiI solution/vapor interface.

Table C.7: Biexponential fitting (2 ps) of $C_2(t)$ for water molecules in the LiI solution. The relative standard errors: $\Delta A/A \leq 10^{-2}$, $\Delta \tau_2/\tau_2 \leq 3 \times 10^{-2}$.

water molecules	A	$\tau_2 \; (\mathrm{ps})$
DD'A	0.84	4.04
$\mathrm{DD'AA}$	0.85	4.08
DDAA	0.87	3.76
D'AA	0.78	0.97

Appendix D

Propensities of ions

D.1 Free energy of the water separated and the contact ion pair

From the blue-moon ensemble method[247, 248], we can obtain the constraint force (unit: a.u.force) acting on the atoms. If the distance between the ion pair (unit: Å) is chosen as the reaction coordinate, the formula for calculating the free energy (unit: kcal/mol) is given as follows. The relative free energy is given by

$$F = \sum_{i}^{N} f_i \Delta r,$$

where i denote a point on the one-dimensional reaction coordinate, N is the number of the sampling points of the reaction coordinates, and f_i denotes the average force on atoms over the trajectory when i is fixed. Now we estimate the error of the free energy δF from the summation approximation. It reads

$$\delta F = \frac{1}{N} \sum_{i}^{N} \delta f_i \Delta r. \tag{D.1}$$

Usually, $\delta f_i \approx \delta f$, thus

$$\delta F = \frac{1}{N} \delta f \sum_{i}^{N} \Delta r. \tag{D.2}$$

Particularly, if $\Delta r = 0.2$ Å, $\delta f = 0.0075$ a.u.force, we get

$$\delta F \approx 1.78 \text{ kcal/mol.}$$

D.2 Surface tension increment

The surface tension increment $d\gamma/dm_2$ is derived as follows. At constant temperature T and constant pressure p, the Gibbs-Duhem equation is [144]:

$$Ad\gamma + n_1^{\sigma} d\mu_1 + n_+^{\sigma} d\mu_+ + n_-^{\sigma} d\mu_- = 0, \tag{D.3}$$

where water is component 1, and the chemical potential for the ionic species i(i = +, -) is defined as

$$\mu_i = \mu_i^0 + RT \ln a_i + z_i F \phi, \tag{D.4}$$

where F is Farady constant, ϕ is the electrical potential of this region (surface or bulk), and z_i is the valence of species i, with ionic activity $a_i = f_i m_i$ in that region, where f_i is the activity coefficient, and m_i is the surface or bulk molality of the ion.

Differentiate Eq. D.3 with respect to m_2 , then the surface tension increment is

$$\frac{d\gamma}{dm_2} = -\frac{1}{m_1^*} \frac{n_1^{\sigma}}{A} [(m_+^{\sigma} - m_+^b) \frac{d\mu_+}{dm_2} + (m_-^{\sigma} - m_-^b) \frac{d\mu_-}{dm_2}], \tag{D.5}$$

where m_1^* is the solvent molality.

For species i, distribution coefficient $K_{p,i}$, is defined as

$$K_{p,i} = \frac{m_i^{\sigma}}{m_i^b} \approx \frac{m_i^{\sigma}}{\nu_i m_2}.$$
 (D.6)

$$\frac{d\mu_{\pm}}{dm_2} = \frac{RT}{m_2} (1 + \epsilon_{\pm}^b) + z_{\pm} F \frac{d\phi^b}{dm_2} \approx \frac{RT}{m_2} (1 + \epsilon_{\pm}^b), \tag{D.7}$$

where $\epsilon_{\pm}^b = d(\ln f_{\pm}^b)/d(\ln m_2)$.

The ion partition coefficients $K_{p,2}$ is defined:

$$\nu K_{p,2} = \nu_+ K_{p,+} + \nu_- K_{p,-}, \tag{D.8}$$

where ν_i is ions' stoichiometry.

In general, from Eq. D.5, Eq. D.6 and Eq. D.7, for solutions at sufficient low electrolyte concentrations, the surface tension increment is

$$\frac{d\gamma}{dm_2} = -\frac{1}{m_1^*} \frac{n_1^{\sigma}}{A} \sum_{i=\pm} (m_i^{\sigma} - m_i^b) \frac{d\mu_i}{dm_2}$$

$$= -\frac{RT\nu}{m_1^*} \frac{n_1^{\sigma}}{A} \sum_{i=\pm} \frac{\nu_i (K_{p,i} - 1)(1 + \epsilon_i^b)}{\nu}$$

$$= -\frac{RT\nu}{m_1^*} \frac{n_1^{\sigma}}{A} [(K_{p,2} - 1) + \sum_{i=\pm} \frac{(K_{p,i} - 1)\nu_i \epsilon_i^b}{\nu}]$$
(D.9)

Under the condition $\epsilon_{\pm}^b << 1$ or $\epsilon_{+}^b \approx \epsilon_{-}^b$ the surface tension increment is

$$\frac{d\gamma}{dm_2} = -\frac{RT\nu}{m_1^*} \frac{n_1^{\sigma}}{A} (1 + \epsilon_i^b)^0 \left[\frac{\nu_+ K_{p,+} + \nu_- K_{p,-}}{\nu} - 1 \right]. \tag{D.10}$$

The approximations in Eq. D.10 regarding the self-interaction nonideality terms ϵ for the salt component, for the single ions, and as a geometric mean of the single ion terms, are given in Ref.[144]. Typically the mean ionic version of ϵ is of order \sim 0.1, so smaller in magnitude than 1. (Thomas Record) Here, nonideality means the activity coefficient f_i is not much smaller than 1.

Appendix E

Thickness of the interface of aqueous solutions

E.1 Interfacial HB dynamics

Instantaneous Surfaces Let us consider the interfacial system of pure water. At a certain time t, the instantaneous surface $\mathbf{s}^0(x(t),y(t))$ can be determined by calculating the coarse-graining density field: we can specify that the coarse-grained density is the reference density $\rho_{\text{ref}} = 0.016 \text{ Å}^{-3}$, and those grid points constitute the surface $\mathbf{s}^0(x,y)$ of pure water. In our simulated pure water system, there are two such surfaces. The code for calculating the instantaneous surfaces of interfacial systems can be found on https://github.com/hg08/interface.

Griding and layering Wee assume that the normal is along the z-axis direction. First, we discretize the coordinates of the xy plane. We divide the edges along the x-axis direction and the edges along the y-axis direction of the simulated box into N parts uniformly, and the xy plane can be approximated by $N \times N$ discrete points. Then $N \times N$ ordinates of the surface $\mathbf{s}^0(x(t_0), y(t_0))$ at the initial time t_0 can be represented as components of an $N \times N$ -vector $\mathbf{z}^0(t_0)$. The surface $\mathbf{s}^0(x(t_0), y(t_0))$ is also the upper boundary of the interface. Secondly, we define a layering strategy, or define the thickness d of the interface layer. In this way, we then determine the lower boundary $\mathbf{s}^1(x(t_0), y(t_0))$ of the interface. It can be expressed as a $N \times N$ -vector $\mathbf{z}^1(t_0) = \mathbf{z}^0(t_0) - \mathbf{d}$, and \mathbf{d} is a $N \times N$ -dimensional constant vector in which all the entries are d. The superscripts 0 and 1 respectively identify the upper and lower boundaries of the interface. Similarly, the other interface can also be determined.

So far, our interfaces are still static, because we only considered the initial moment. But the instantaneous interfaces can be defined for different moments naturally. Therefore, according to the same method, for the molecular motion trajectory of an interface system, we can obtain the dynamically changing interface layer, i.e., $\mathbf{z}^1(t) = \mathbf{z}^0(t) - \mathbf{d}$. When the interfaces are determined, we can answer the question: For an atom with coordinates (x, y, z) at any time t, whether it is in the interface? To answer this question, we map the atom's coordinates (x, y) to an ordered integer pair $(i, j), i, j \in \{0, \dots, N\}$, where $i = \operatorname{int}(x/\Delta x), j = \operatorname{int}(y/\Delta y), \Delta x = a/N$, and $\Delta y = b/N$.

At a certain time t, the ordinate z of an atom can be approximately represented as a function defined on point (i,j), z=z(i,j). Then we compare z(i,j) and $z_{i,j}^l$, l=0,1 to determine if the atom is in the interface. That is, if $z_{i,j}^0(t) < z(i,j) < z_{ij}^1(t)$, then the atom at (x,y,z) is located in the interface at time t.

Molecule Sampling (MS) from instantaneous interface The method of Molecule Sampling used to extract the interfacial molecules is as follows. 1) Determine the instantaneous surface of the water/vapor interface system; 2) Define a interfacial layer with a fixed thickness d below the surface; 3) Select the atoms which are near the instantaneous surface; 4) Calculate the HB dynamics for molecules in interface at different time t. Given the thickness of layer d, at any sampling time $t_0, t_1, t_2, \cdots, t_n$, the set of molecules in the interface can be determined. Since the molecules are always in motion, generally, the set $S^0(t_i)$ of the molecules in the interface at time t_i and the set $S^0(t_j)$ of the molecules in the interface at a different time t_j is different. Therefore, to calculate the HB dynamics of molecules in the interface, we calculate the correlation functions of the HB population h(t) of the molecules in the interface at different times, and calculate the average of the obtained correlation functions over the n functions, e.g., $C_{\text{HB},j}(t), j = 1, \cdots, n$.

Interfacial hydrogen bond population operator $h^{(s)}(t)$ The IHB method to extract the interfacial molecules is as follows. 1) Determine the instantaneous surface of a system; 2) Define a interface system with a fixed thickness below the instantaneous surface; 3) Define the interfacial hydrogen bond population operator $h^{(s)}(t)$. 4) Calculate the autocorrelation function of $h^{(s)}(t)$ and then the related observations.

In chapter 5, we had combined the recognition technology of the instantaneous liquid interfaces [74] with the definition of H-bonds [182, 214] to define H-bonds that depend on the environment. With this method, we can have more understanding of the differences and commonalities in the kinetics of the breaking and reconstruction of the hydrogen bonds of the water molecules in the interface system. At the same time, we can compare the HB dynamics in the interface of different thicknesses with the HB dynamics in bulk water to obtain the thickness of the interface from the perspective of HB dynamics. In addition, for the interface system of different solutions, we can also use the layering technique to study the HB dynamics, so that we can understand the particularity of the interface of various solutions relative to pure water.

None of the two methods (MS and IHB) for obtaining the HB dynamics of the interface water molecules can give the true interface HB dynamics completely and accurately. But they respectively give an extreme case of interface HB dynamics. In the MS method, the formation and breaking of intermolecular H-bonds can be truly described, but the selection of interface water molecules is not accurate enough. Since the configuration of the molecule will change over time, the contribution of the H-bonds in the bulk phase will be included. In the IHB method, we are very accurate in choosing the interface water molecules and the H-bonds in the interface, but we may artificially destroy some H-bonds in the interface that were not broken. To a certain extent, the HB dynamics obtained by the IHB method is accelerated. Therefore, the comparison of the results obtained by these two methods may give a true picture of the interface HB dynamics. In particular, as the interface thickness increases, the two methods can achieve the same results. The code for calculating the HB dynamics for instantaneous interfaces can be found on https://github.com/hg08/hb_in_interface.

E.2 Thickness of the solution/vapor interfaces

To determine the thickness of the water/vapor interface of alkali solutions, we chose several different thickness values of slab of the interface and calculate the corresponding susceptibility for these slabs, respectively. Take the water/vapor interface of

LiNO₃ solution as an example. We chose several different thickness values—from 2 to 8 Å, and calculate VSFG intensities $I_{SSP} \propto |\chi_{SSP}^{(2),\mathrm{R}}|^2$ for the water/vapor interface with a thickness of each of these values. The result is given in Fig. E.1 a and b. It shows that $|\chi_{SSP}^{(2),\mathrm{R}}|^2$ converges as the thickness increases.

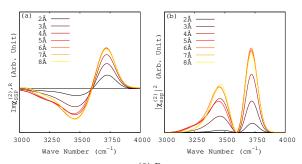


FIGURE E.1: The calculated $|\chi_{SSP}^{(2),R}|^2$, of water molecules at the aqueous/vapor interfaces with different thickness.

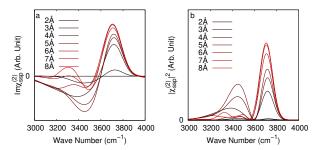


FIGURE E.2: The calculated $|\chi_{SSP}^{(2),R}|^2$, of water molecules at the interfaces of the LiI solution with different thickness.

The VSFG intensities $I_{SSP} \propto |\chi_{SSP}^{(2),R}|^2$ for the water/vapor interface of the LiI solution are given in Fig. E.2 a and b. The results show that $|\chi_{SSP}^{(2),R}|^2$ increases as the thickness increase from 0 to 8 Å.

Furthermore, we also estimated the thickness of a water/vapor interface by calculating the HB dynamics for it. Fig. E.3 shows the dependence of the logarithm of survival probability on the thickness of the water/vapor interface of the LiI solution. We found that when d=8 Å the correlation function $\ln S_{\rm HB}(t)$ of the interface converges as the thickness increases, indicating the thickness of the interface.

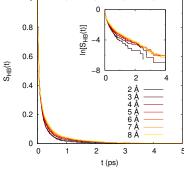


FIGURE E.3: The logarithms of the function $S_{\rm HB}(t)$ for water-water H-bonds at interfaces with different thickness in the LiI solution.

In addition, the survival probability $S_{\rm HB}(t)$ is dependent on the temporal resolution $t_{\rm t}$, which is the time interval between two adjacent states in time used to calculate the survival probability. As an example, the $t_{\rm t}$ dependence of the $\tau_{\rm HB}$ of the three alkali-iodine solution interfaces in AIMD simulations is reported in Fig. E.4. It shows that, if one take the $t_{\rm t}$ as small as possible, the number of times that H-bonds break and form again in this period of time $(t_{\rm t})$ will be reduced. As shown in Fig. E.4, The value of $\tau_{\rm HB}$ corresponding to the intersections of the $\tau_{\rm HB}(t_{\rm t})$ functions and the line $t_{\rm t}=5$ fs approximately give the continuum HB lifetimes. For $t_{\rm t}=5$ fs, the calculated continuum HB lifetime is 0.30, 0.31 and 0.23 ps, for the interface of LiI, NaI and KI solution, respectively. Then, the estimated value of $\tau_{\rm a}$ is ~ 0.2 ps. Moreover, we can obtain the continuum HB lifetime independent of the $t_{\rm t}$: $\tau_{\rm HB}=\lim_{t_{\rm t}\to 0}\tau_{\rm HB}(t_{\rm t})$.

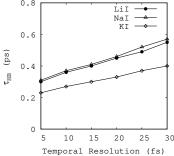


FIGURE E.4: The resolution dependence of the continuum lifetime $\tau_{\rm HB}$ of water–water H-bonds at interfaces of different alkali-iodine solutions at 330 K, calculated for six temporal resolutions $(t_{\rm t})[249-251]$.

Appendix F

HB dynamics and instantaneous interfaces

F.1 Relations between HB lifetime distributions

The HB lifetime is a significant feature of the HB dynamics in liquids. The peculiar properties of water are a direct consequence of water's H-bond lifetime distribution.[196, 203, 252] For example, understanding of HB dynamics is essential when investigating proton transfer reactions in protein environments.[253] The statistical properties of lifetime of H-bonds can be described by a variety of distribution functions.[199, 207, 254, 255] Below we discuss the three probability distribution functions of HB lifetime, so that we can study the dynamic characteristics of H-bonds in liquids and interfaces.

Probability distribution of HB lifetimes in a configuration Suppose that there are n_{tot} H-bonds in a system at time t, and we can distinguish a part of H-bonds from these H-bonds. The lifetime of this part of H-bonds is in a certain range $[\tau, \tau + d\tau]$. We can assume that their number of those bonds is n_{τ} . One can easily find that $n_{\text{tot}} > n_{\tau}$. If we observe this part of H-bonds in the next time period $[t, t + \tau]$, then they will be broken once during $[t, t + \tau]$. That is to say, within $[t, t + \tau]$, we will detect the breaking of all H-bonds with lifetime τ (see Fig. F.1). Therefore, in a very short period of time $d\tau$, the probability of detecting these H-bonds is $(1/\tau)d\tau$. Since the probability for the HB to have the lifetime $\in [\tau, \tau + d\tau]$ is $P_{t}(\tau)$. Therefore, the relation between $P_{a}(t)$ and $P_{t}(t)$ is

$$P_{\rm a}(t) = \int_{t}^{\infty} P_{\rm t}(\tau) \frac{d\tau}{\tau},\tag{F.1}$$

i.e., the probability of the HB breaking for the first time in the time t after detection at the initial moment depends on the number of those H-bonds whose lifetime exceeds the given time t. [206]

Different HB lifetime distributions From the probability $P_{\rm tc}(t)$ of the total HB lifetime in a configuration, and the probability $P_{\rm a}$ of the first HB breaking in time t after it have been detected at the moment t, one can introduce the average time $\langle \tau_{\rm tc} \rangle$ and $\langle \tau_{\rm a} \rangle$:

$$\langle \tau_{\rm tc} \rangle = \int_0^\infty t P_{\rm tc}(t) dt,$$
 (F.2)

$$\langle \tau_{\rm a} \rangle = \int_0^\infty t P_{\rm a}(t) dt.$$
 (F.3)

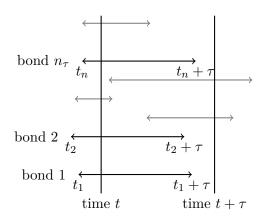


FIGURE F.1: The H-bonds with lifetime τ in a certain configuration. At time t, we assume that there are totally $n_{\rm tot}$ H-bonds can be detected, and n_{τ} H-bonds are of lifetime τ , therefore, the fraction of H-bonds that have the lifetime τ in the configuration at time t is $P_{\rm tc}(\tau) = n_{\tau}/n_{\rm tot}$. Let τ take all the values in the interval $[0, \infty]$, we can get the HB lifetime distribution $P_{\rm tc}(t)$.

Since $P_{\rm a}(t) = \int_t^{\infty} P_{\rm tc}(\tau) \frac{d\tau}{\tau}$, i.e.,

$$P_{\rm tc}(t) = -t \frac{dP_{\rm a}(t)}{dt},$$

integrating by parts, we obtain

$$\langle \tau_{\rm tc} \rangle = -\int_0^\infty t^2 \frac{dP_{\rm a}(t)}{dt} dt$$
$$= -\int_0^\infty t^2 dP_{\rm a}(t)$$
$$= 2\langle \tau_{\rm a} \rangle,$$

in which we used $\int_0^\infty d(t^2 P_a) = 0$. Therefore, there is a relationship between $\langle \tau_{tc} \rangle$ and $\langle \tau_a \rangle$:

$$\langle \tau_{\rm tc} \rangle = 2 \langle \tau_{\rm a} \rangle.$$
 (F.4)

We denote the probability of the total HB lifetime along a trajectory as $P_{\rm tt}(t)$, then the average HB lifetime over the trajectory is

$$\langle \tau_{\rm tt} \rangle = \int_0^\infty t P_{\rm tt}(t) dt.$$
 (F.5)

Because $\int_0^\infty P_{\rm tc}(t)dt = \frac{1}{\langle \tau_{\rm tt} \rangle} \int_0^\infty t P_{\rm tt}(t)dt = 1$, we get

$$P_{\rm tt}(t) = \langle \tau_{\rm tt} \rangle P_{\rm tc}(t)/t.$$
 (F.6)

The difference between the two distribution functions, $P_{\rm tt}(t)$ and $P_{\rm tc}(t)$, can be described as follows. The $P_{\rm tt}(t)$ represents the percentage of pairs of molecules that had a continuous H-bonds during time t, while the $P_{\rm tc}(t)$ the percentage of the number of H-bonds with a given lifetime t to the number of all H-bonds in any configuration. [206]

Since $P_{\rm a}(t) = \int_t^{\infty} P_{\rm tc}(\tau) \frac{d\tau}{\tau}$, we can obtain

$$P_{a}(t) = \int_{t}^{\infty} \frac{P_{tt}}{\langle \tau_{tt} \rangle} \frac{\tau}{\tau} d\tau$$
$$= \int_{t}^{\infty} \frac{P_{tt}}{\langle \tau_{tt} \rangle} d\tau.$$

Let t = 0, we obtain

$$P_{\rm a}(0) = 1/\langle t_{\rm tt} \rangle = 1/\langle t_{\rm HB} \rangle$$
. (F.7)

From Eq. F.2 and the relation between $S_{\rm HB}(t)$ and $P_{\rm a}(t)$

$$S_{\rm HB}(t) = \int_{t}^{\infty} P_{\rm a}(\tau) d\tau, \tag{F.8}$$

we can obtain

$$\begin{split} &\int_0^\infty \int_t^\infty P_{\mathbf{a}}(t) d\tau dt = \int_0^\infty \int_0^\tau P_{\mathbf{a}}(\tau) dt d\tau \\ &= \int_0^\infty \tau P_{\mathbf{a}}(\tau) d\tau \\ &= \langle \tau_{\mathbf{a}} \rangle, \end{split}$$

i.e.,

$$\int_0^\infty S_{\rm HB}(t)dt = \langle \tau_{\rm a} \rangle. \tag{F.9}$$

Calculation of HB lifetime distributions In this paragraph, we describe the method to calculate the above lifetime distributions $P_{tc}(\tau)$, $P_{a}(\tau)$, and $P_{tt}(\tau)$.

First, we describe the method of calculating $P_{\rm tc}(\tau)$. Theoretically speaking, in order to calculate $P_{\rm tc}(\tau)$, our detection time t must meet the following conditions: $t - t_0 > \tau_{\rm hb}^{\rm max}$, where t_0 is the initial time and $\tau_{\rm hb}^{\rm max}$ is the maximum lifetime value of the H-bonds in the system. However, the value of $\tau_{\rm hb}$ cannot be known in advance. In order to reduce the error, the method we can adopt is to set an empirical value as large as possible for $\tau_{\rm hb}^{\rm max}$ if conditions permit. Since the value of $\tau_{\rm hb}^{\rm max}$ is limited, in principle the lifetime value of the HB can always be greater than $\tau_{\rm hb}^{\rm max}$. Therefore, the average value of the HB lifetime distribution calculated in this approximate way will move to a shorter lifetime than the average value of the true HB lifetime distribution:

$$\int_0^\infty \tau P_{\rm tc}^{\rm approx}(\tau) d\tau < \langle \tau_{\rm tc} \rangle. \tag{F.10}$$

Among the H-bonds detected at time t, if there are some H-bonds that have existed at the beginning t_0 and remain in existence until time t, then we can approximately express the lifetime of these H-bonds as $\delta t^{(j)} = t^{(j)} - t_0$, where $j = 1, \dots, m$ is the labels of the m H-bonds and $t^{(j)}$ is the moment when the j-th HB is broken. If we use $\tau^{(j)}, j = 1, \dots, m$ to represent the true lifetimes of these m H-bonds, then we can find that $\tau^{(j)} - \delta t^{(j)} > 0$. Since we cannot judge the true lifetime of these m hydrogen bonds, we can use $\delta t^{(j)}$ to approximate the lifetime of these m H-bonds, that is

$$\tau^{(j)} = \delta t^{(j)}.\tag{F.11}$$

For those H-bonds that did not exist at the beginning, the method of calculating their lifetime is very straightforward, the lifetime $\tau^{(j)}$ is equal to the time $t^{(j)}$ when the HB is broken minus the moment $t_0^{(j)}$ of its formation:

$$\tau^{(j)} = t^{(j)} - t_0^{(j)}, \tag{F.12}$$

where the superscript $j=1,\dots,m'$, identifies m' H-bonds detected at time t, and formed after t_0 and broken at $t^{(j)}$.

Specifically, for the DFTMD simulation results we obtained, we also approximate P_{tc} as follows: We select evenly distributed n time points $t = t_1, ..., t_n$, from the trajectory obtained by the simulation, and count the HB lifetimes at each time point t_i . The distribution function $P_{tc}(\tau)$ can be obtained by the average of the lifetime distribution detected at a certain time t_i , $i = 1, \dots, n$, where $t_i - t_{i-1} = \tau_{bb}^{max}$.

If only from the perspective of simulation data, we have another way to obtain $P_{\rm tc}(\tau)$: Count the lifetimes of H-bonds that are formed after the initial time t_0 and are broken before the end time $t_{\rm f}$. Although the distribution obtained in this method cannot be verified experimentally, it is the true distribution of HB bonds in simulated systems.

F.2 Hydrogen bond population operator

Calculation of the reactive flux For dynamical variables $x_i(t)$ and $x_k(t)$, their correlation functions have the following relationship: [256]

$$\langle x_i(t')x_k(t)\rangle = -\langle x_i(t)x_k(t')\rangle,$$
 (F.13)

or

$$\phi_{ik}(t) = -\phi_{ki}(t).$$

Let $x_i = h$, $x_k = \dot{h}$, then, we obtain

$$\langle h(0)\dot{h}(t)\rangle = -\langle \dot{h}(0)h(t)\rangle.$$
 (F.14)

From the definition of the reactive flux k(t) = -dc/dt, we obtain

$$k(t) = -\langle h(0)\dot{h}(t)\rangle/\langle h\rangle. \tag{F.15}$$

Then from Eq. F.14, we get

$$k(t) = \langle \dot{h}(0)h(t)\rangle/\langle h\rangle.$$

Since $\langle \dot{h}(0) \rangle = 0$, k(t) can be calculated by

$$k(t) = -\langle \dot{h}(0)[1 - h(t)] \rangle / \langle h \rangle. \tag{F.16}$$

Relaxation time of hydrogen bonds in bulk water For the dynamic trajectory of such a system, we also calculated the self-correlation function c(t) of the HB population operator, and the functions k(t) and n(t) derived from it. Table F.1 and F.2 shows the rate constant k, k' and the relaxation time $\tau_{\rm HB}$ obtained by the least squares fit method. It can be seen from the tables that the accuracy of the calculation of k, k' are accurate to at least two decimal places.

TABLE F.1: The k and k' for the bulk water. We carried on the short time region 0.2 ps < t < 2 ps. The unit for k (k') is ps⁻¹, and that for τ_{HB} (= 1/k) is ps. The h(t) is bond-based.

Criterion	k (bulk)	k' (bulk)	τ_{HB} (bulk)
ADH	0.299	1.029	3.347
AHD	0.288	1.121	3.468

TABLE F.2: The k and k' for the bulk water. We carried on the long time region 2 ps < t < 12 ps. The unit for k (k') is ps⁻¹, and that for $\tau_{\rm HB}$ (= 1/k) is ps. The h(t) is bond-based.

Criterion	k (bulk)	k' (bulk)	$ au_{ m HB}~(m bulk)$
ADH	0.103	0.028	9.728
AHD	0.103	0.040	9.702

F.3 Reorientation dynamics of water molecules at solution/vapor interfaces

Double exponential fit In general, the rotational motion of water molecules are not simply characterized by a well-defined rate constant. Similar non-single-exponential kinetics is also obtained in the HB kinetics in liquid water [192, 257] and in the time variation of the average frequency shifts of the remaining modes after excitation in hole burning technique. [258, 259] We can understand the non-single-exponential kinetics of rotational anisotropy decay by fitting the rotational anisotropy decay by a biexponential function.

To obtain the effects of diffusion and HB decay of water molecules in solutions respectively, we assume that the correlation function $C_2(t)$ has the form [260]

$$C_2(t) = A_1 e^{-t/\tau_{2,1}} + A_2 e^{-t/\tau_{2,2}},$$
 (F.17)

where A_i are amplitudes and $1/\tau_{2,i}$ (i=1,2) are decay rates of $C_2(t)$, and the $\tau_{2,i}$ represent the relaxation times of the $C_2(t)$. The form of this expression means: We assume that the orientation relaxation $C_2(t)$ is composed of two independent relaxation processes with different time scales, and they can be described by an exponential decay function, respectively. For alkali-iodine solutions (LiI and NaI), the A_i and $\tau_{2,i}$ of the biexponentials fits for functions $C_2(t)$ are shown in Table F.3. We found that Eq. F.17, the biexponential kinetics is sufficiently accurate to describe the characteristics of anisotropy decay.

Then we considered the effect of ion species in solution/vapor interfaces on the anisotropy decay of water molecules. From Table F.3, we found that for both LiI and NaI solutions, there are two decay processes — amplitude ~ 1 , decay time $\tau_{2,2} \sim 10.0$ ps, and for the other describe the initial fast decay of the anisotropy, with amplitude ~ 0.1 , decay time $\tau_{2,1} \sim 0.1$ –1.0 ps, due to the inertial–librational motion preceding the orientational diffusion.

TABLE F.3: The fitted parameters of anisotropy decay of water molecules at the LiI (NaI, KI) solution/vapor interface according to Eq. F.17 for the time range [0,10] ps. The standard errors: $\delta A_i \sim 0.01$, $\delta \tau_{2,i} \sim 0.1$ ps (i=1,2).

H_2O	A_1	$\tau_{2,1} \; ({\rm ps})$	A_2	$\tau_{2,2} \; ({\rm ps})$
Li ⁺ -shell	0.56	8.3	0.33	50.0
I^- -shell	0.86	7.1	0.08	0.1
Na ⁺ -shell	0.71	16.7	0.19	1.3
LiI/vapor	0.84	4.9	0.09	0.6
NaI/vapor	0.77	9.1	0.13	0.4
KI/vapor	0.80	6.1	0.12	1.1

The orientation relaxation time constants $\tau_{2,i}$ and amplitudes A_i (i=1,2) of the single exponentials fits (10 ps) for $C_2(t)$ are in Table F.3. For comparison, we also fit the $C_2(t)$ function of the water molecules in each interface in Table 1 with a single exponential. The results are shown in table F.4. It can be seen that the value of τ_2 is between the two characteristic times $\tau_{2,1}$ and $\tau_{2,2}$, obtained by biexponential fitting. At the same time, we can see the obvious difference in water molecules under various conditions: the reorientation characteristic time τ_2 in the first solvation shell of alkali metal ions is 2 to 3 times that of I⁻ ion. This is a reasonable result. However, the single exponential model describes the reorientation of water molecules as an exponential decay process. Our above biexponential function model (Eq. F.17) can be explained as: the reorientation dynamics of each water molecule is composed of two independent decay processes. One has the amplitude ~ 1 , decay time ~ 10 ps, which describes the initial fast decay of the anisotropy, and the other is of amplitude ~ 0.1 , decay time ~ 0.1 ps, due to the inertial-librational motion preceding the orientational diffusion.

Table F.4: The fitted (single exponential) parameters of anisotropy decay of water molecules at the LiI (NaI, KI) solution/vapor interface according to Eq. F.17 for the time range [0,10] ps. The standard errors: $\delta A \sim 0.01, \, \delta \tau_2 \sim 0.1 \,$ ps.

H_2O	A	$\tau_2 \; (\mathrm{ps})$
Li ⁺ -shell	0.88	13.7
I^- -shell	0.83	3.9
Na ⁺ -shell	0.84	10.1
LiI/vapor	0.80	3.0
NaI/vapor	0.86	4.2
KI/vapor	0.86	5.7

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