# Introduction to Solid State Physics (C. Kittel)

**Chapter 2** 

## WAVE DIFFRACTION AND THE RECIPROCAL LATTICE

## **Contents**

## DIFFRACTION OF WAVES BY CRYSTALS The Bragg law

#### SCATTERED WAVE AMPLITUDE

Fourier analysis
Reciprocal lattice vectors
Diffraction conditions
Laue equations

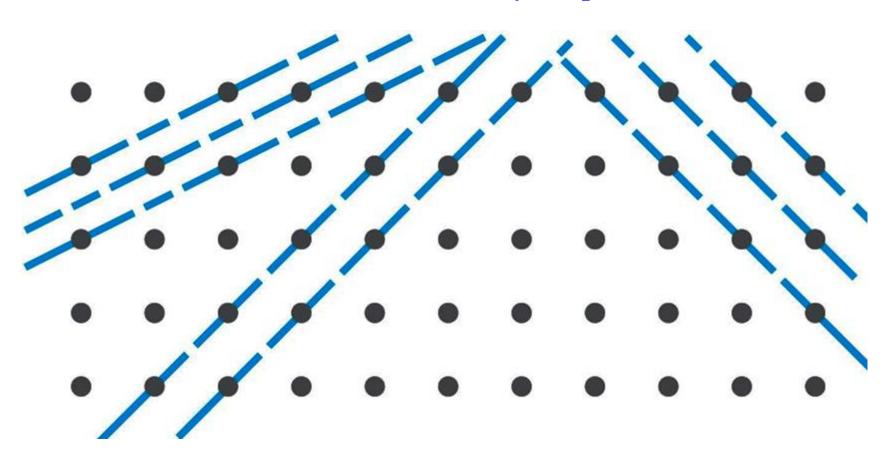
#### BRILLOUIN ZONES

Reciprocal lattice to sc lattice Reciprocal lattice to bcc lattice Reciprocal lattice to fcc lattice

#### FOURIER ANALYSIS OF THE BASIS

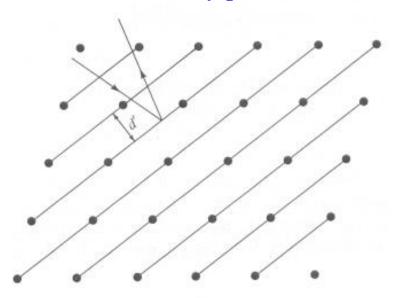
Structure factor of the bcc lattice Structure factor of the fcc lattice Atomic form factor

## different crystal planes



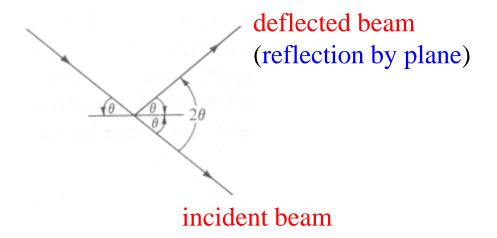
The Bragg law: diffraction condition in terms of reflection by planes

reflection by plane



#### Bragg angle $\theta$

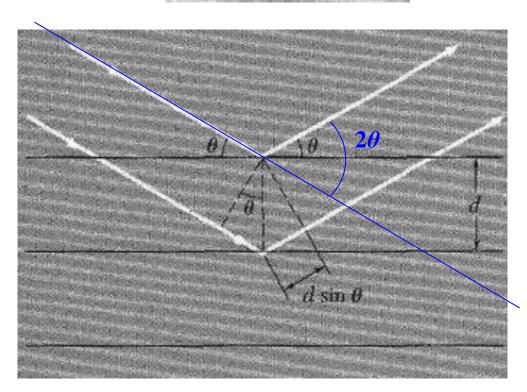
: the half of the total angle  $2\theta$  by which the incident beam is deflected



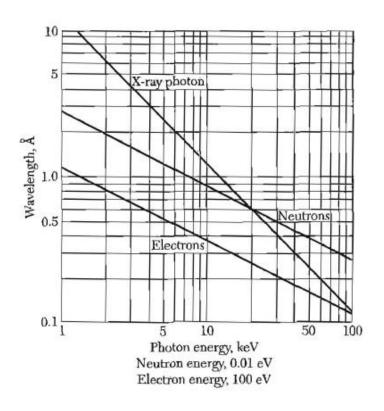
The Bragg law Bragg condition for diffraction

$$2d \sin \theta = n\lambda$$

only for wavelength  $\lambda \leq 2d$ .



## The Bragg law



**Bragg condition for diffraction** 

 $\lambda \sim d$   $d \sim 0.1 - 1$  nm

## Quiz

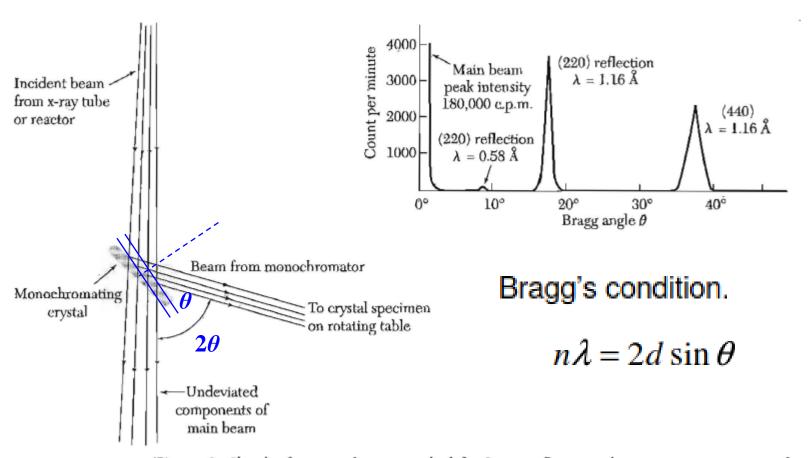
With  $d \sim 1$  nm, calculate the beam energy (in eV) of each prove beam for diffraction by crystals.

- (1) Light (EM wave)
- (2) Electron
- (3) neutron

Hint: From Bragg condition for diffraction

Observability condition:  $\lambda \sim d$ 

#### The Bragg law (application: selection of monochromatic beam)



**Figure 3** Sketch of a monochromator which by Bragg reflection selects a narrow spectrum of x-ray or neutron wavelengths from a broad spectrum incident beam. The upper part of the figure shows the analysis (obtained by reflection from a second crystal) of the purity of a 1.16 Å beam of neutrons from a calcium fluoride crystal monochromator. (After G. Bacon.)

The Bragg law (application: selection of monochromatic beam)

Bragg's condition.  $n\lambda = 2d \sin \theta$ 

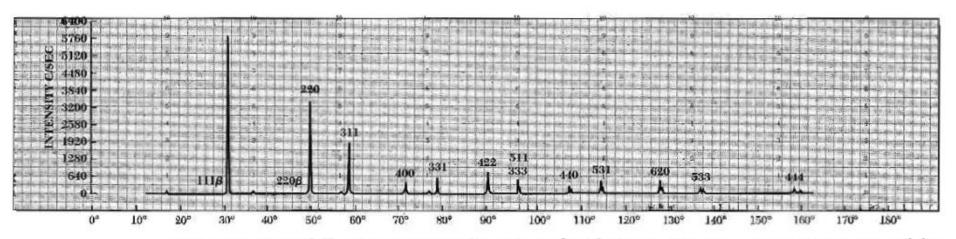
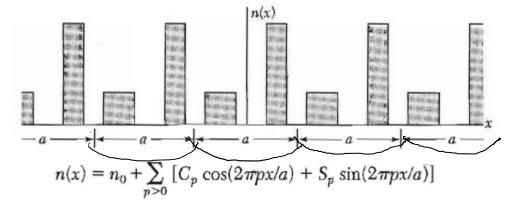


Figure 4 X-ray diffractometer recording of powdered silicon, showing a counter recording of the diffracted beams. (Courtesy of W. Parrish.)

## Fourier analysis

$$n(x+a) = n(x)$$
  $T = na$ 

Real space



1-dimensional example

$$n(x + a) = n_0 + \sum [C_p \cos(2\pi px/a + 2\pi p) + S_p \sin(2\pi px/a + 2\pi p)]$$
  
=  $n_0 + \sum [C_p \cos(2\pi px/a) + S_p \sin(2\pi px/a)] = n(x)$ .

$$n(x) = \sum_{p} n_p \exp(i2\pi px/a)$$
 : real if  $n_{-p}^* = n_p$  is satisfied

## Fourier analysis

#### In 1-dimension

$$n(x) = \sum_{p} n_p \exp(i2\pi px/a)$$
 : real if  $n_{-p}^* = n_p$  is satisfied

 $2\pi/a$  is a point in the reciprocal lattice or Fourier space of the crystal.



$$n(\mathbf{r} + \mathbf{T}) = n(\mathbf{r})$$
  $\mathbf{T} = u_1 \mathbf{a}_1 + u_2 \mathbf{a}_2 + u_3 \mathbf{a}_3$ 

$$n(\mathbf{r}) = \sum_{\mathbf{G}} n_{\mathbf{G}} \exp(i\mathbf{G} \cdot \mathbf{r})$$
  $n_{\mathbf{G}} = V_c^{-1} \int_{\text{cell}} dV \, n(\mathbf{r}) \, \exp(-i\mathbf{G} \cdot \mathbf{r})$ 

G = reciprocal lattice vector in 3D

## **Reciprocal lattice vectors**

 $a_i$ : real-space lattice vector

 $b_i$ : reciprocal space lattice vector

#### **Construction of reciprocal lattice vectors**

$$b_1 = 2\pi \frac{a_2 \times a_3}{a_1 \cdot (a_2 \times a_3)}, \quad b_2 = 2\pi \frac{a_3 \times a_1}{a_1 \cdot (a_2 \times a_3)}, \quad b_3 = 2\pi \frac{a_1 \times a_2}{a_1 \cdot (a_2 \times a_3)}.$$

 $\rightarrow$  If  $\mathbf{a_i}$  are primitive,  $\mathbf{b_j}$  are primitive.

**Reciprocity condition** 

$$\mathbf{b}_i \cdot \mathbf{a}_j = 2\pi \delta_{ij}$$

## **Reciprocal lattice vectors**

 $a_i$ : real-space lattice vector

 $b_i$ : reciprocal space lattice vector

$$\mathbf{b}_{i} \cdot \mathbf{a}_{j} = 2\pi \delta_{ij}$$
 (reciprocity condition)

 $u_i, v_i$ : integers

$$T = u_1a_1 + u_2a_2 + u_3a_3$$
. Translational vector in real space

$$\mathbf{G} = v_1 \mathbf{b}_1 + v_2 \mathbf{b}_2 + v_3 \mathbf{b}_3$$
$$\exp(i\mathbf{G} \cdot \mathbf{T}) = 1$$

**G**: Translational vector

in reciprocal lattice space

$$\exp(i\mathbf{G} \cdot \mathbf{T}) = \exp[i(v_1\mathbf{b}_1 + v_2\mathbf{b}_2 + v_3\mathbf{b}_3) \cdot (u_1\mathbf{a}_1 + u_2\mathbf{a}_2 + u_3\mathbf{a}_3)]$$

$$= \exp[i2\pi(v_1u_1 + v_2u_2 + v_3u_3)] = \mathbf{1}$$

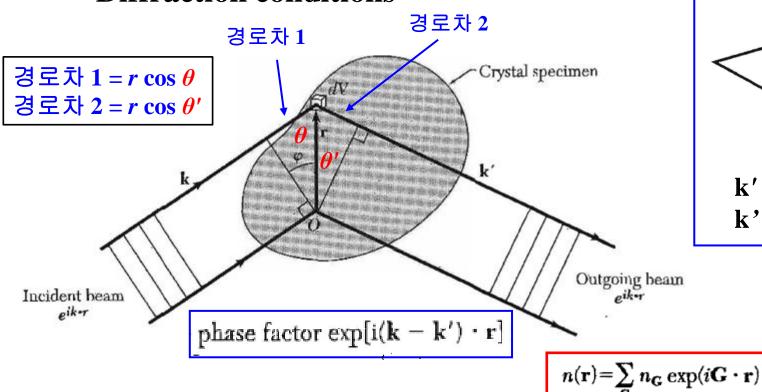
$$n(\mathbf{r} + \mathbf{T}) = \sum_{\mathbf{G}} n_{\mathbf{G}} \exp(i\mathbf{G} \cdot \mathbf{r}) \exp(i\mathbf{G} \cdot \mathbf{T})$$

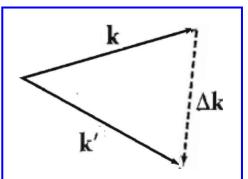
$$n(\mathbf{r} + \mathbf{T}) = n(\mathbf{r}) = \sum n_{\mathbf{G}} \exp(i\mathbf{G} \cdot \mathbf{r}).$$

 $\rightarrow$  G: non-zero k in Fourier expansion

경로차 = 경로차 1 + 경로차 2위상차 = 위상차 1 +위상차 2=  $k (r \cos \theta + r \cos \theta')$ =  $k \cdot r - k' \cdot r$ 

#### Diffraction conditions





$$\mathbf{k'} - \mathbf{k} = \Delta \mathbf{k}$$
, or  $\mathbf{k'} + \Delta \mathbf{k} = \mathbf{k}$ 

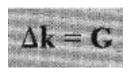
## **Scattering amplitude**

$$F = \int d\mathbf{V} \, n(\mathbf{r}) \, \exp[i(\mathbf{k} - \mathbf{k}') \cdot \mathbf{r}] = \int d\mathbf{V} \, n(\mathbf{r}) \, \exp(-i\Delta \mathbf{k} \cdot \mathbf{r}) = \sum_{\mathbf{G}} \int d\mathbf{V} \, n_{\mathbf{G}} \, \exp[i(\mathbf{G} - \Delta \mathbf{k}) \cdot \mathbf{r}]$$

= 0 unless  $\Delta k = G$ 

$$F = Vn_G$$
 for  $\Delta k = -G$ 

#### Bragg condition



#### **Diffraction condition**



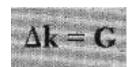
$$d(hkl)$$
 = spacing between parallel lattice planes,  
normal to the direction G (G =  $hb_1 + kb_2 + lb_3$ )  
=  $2\pi/|G|$ 

$$2kG \sin \theta = G^2$$
  $2(2\pi/\lambda) \sin \theta = 2\pi/d(hkl)$   $2d(hkl) \sin \theta = \lambda$   $2d \sin \theta = \lambda$ 

**Laue Equations** 

#### Simple geometrical interpretation

 $\Delta \mathbf{k}$  lies on a certain cone about the direction of  $\mathbf{a}_1$ 





$$\mathbf{a}_1 \cdot \Delta \mathbf{k} = 2\pi v_1$$
;  $\mathbf{a}_2 \cdot \Delta \mathbf{k} = 2\pi v_2$ ;  $\mathbf{a}_3 \cdot \Delta \mathbf{k} = 2\pi v_3$ 

$$\mathbf{a}_2 \cdot \Delta \mathbf{k} = 2\pi v_2$$

$$\mathbf{a}_3 \cdot \Delta \mathbf{k} = 2\pi v_3$$

**Diffraction condition** 

#### **Laue equations**

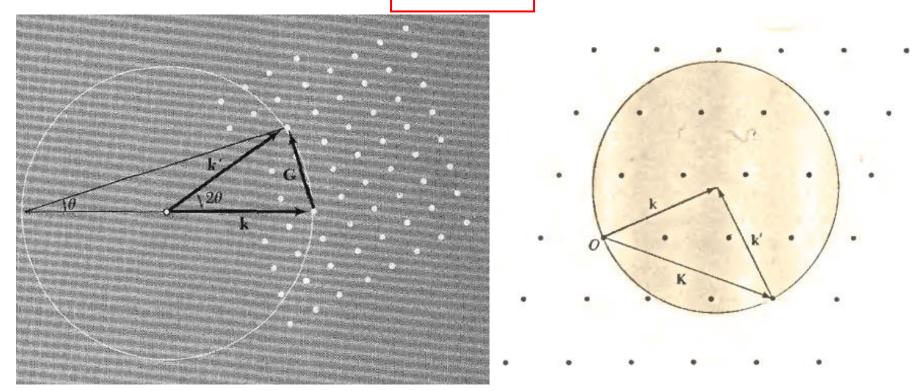
Remember 
$$\mathbf{G} = v_1 \mathbf{b}_1 + v_2 \mathbf{b}_2 + v_3 \mathbf{b}_3$$
  $\mathbf{b}_i \cdot \mathbf{a}_j = 2\pi \delta_{ij}$ 

#### **Laue equations**

- → <u>severe condition</u> to satisfy!
- $\rightarrow$  requires systematic sweeping or searching in  $\lambda$  or crystal orientation (or by sheer accident)

## **Ewald construction**

 $\Delta \mathbf{k} = \mathbf{G}$ 



임의의 En, 임의의 파동 입사 (또는 결정샘플 방향)에 대하여 회절조건을 만족할 수 없다.

Ewald sphere의 <u>반지름 ( $\rightarrow$  파동 En)</u>, <u>파동 입사방향 (또는 결정샘플 방향)</u>을 바꿈으로써 회절 관측 가능!

## **Brillouin Zone**

$$\vec{k} - \vec{k}' = \vec{G}$$

#### **Diffraction condition**

$$\vec{k}' = \vec{k} - \vec{G}$$

$$\left|\vec{k}\right| = \left|\vec{k}'\right| = \left|\vec{k} - \vec{G}\right|$$
 Squaring both sides  $k^2 = k^2 + G^2 - 2\vec{k} \cdot \vec{G}$ 

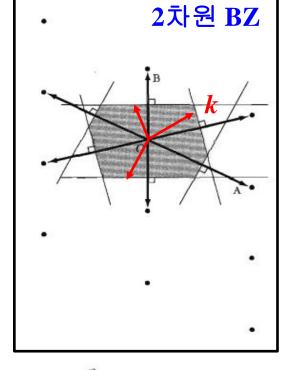
$$\mathcal{K} = \mathcal{K} + \mathcal{K}$$

$$G^2 = 2\vec{k} \cdot \vec{G}$$

or 
$$\frac{G}{2} = \vec{k} \cdot \frac{\vec{G}}{G}$$

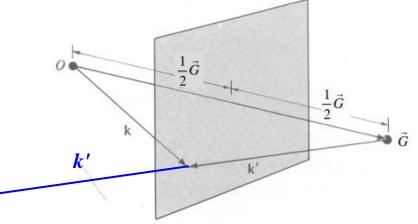
$$2\mathbf{k} \cdot \mathbf{G} + G^2 = 0$$

$$\mathbf{k} \cdot \left(\frac{1}{2} \mathbf{G}\right) = \left(\frac{1}{2} G\right)^2$$



$$\frac{G}{2} = \vec{k} \cdot \frac{\vec{G}}{G}$$

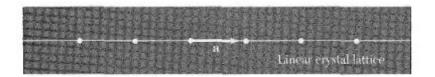
k at "Brillouin Zone boundary"

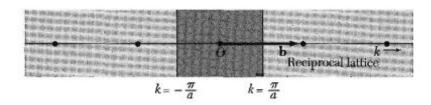


## **Brillouin Zone**

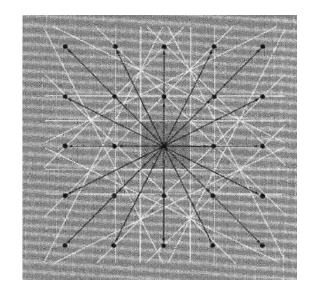
## **Brillouin zones:** Wigner-Seitz cells in k-space

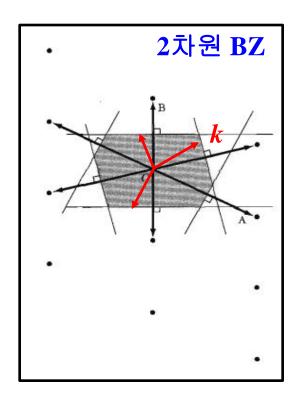
1차원





2차원





#### Reciprocal Lattice to bcc Lattice

The primitive translation vectors of the bcc lattice (Fig. 12) are

$$\mathbf{a}_1 = \frac{1}{2} a(-\hat{\mathbf{x}} + \hat{\mathbf{y}} + \hat{\mathbf{z}}) \; ; \quad \mathbf{a}_2 = \frac{1}{2} a(\hat{\mathbf{x}} - \hat{\mathbf{y}} + \hat{\mathbf{z}}) \; ; \quad \mathbf{a}_3 = \frac{1}{2} a(\hat{\mathbf{x}} + \hat{\mathbf{y}} - \hat{\mathbf{z}}) \; ,$$

$$V = |\mathbf{a}_1 \cdot \mathbf{a}_2 \times \mathbf{a}_3| = \frac{1}{2} a^3$$

$$V_k = |\mathbf{b}_1 \cdot \mathbf{b}_2 \times \mathbf{b}_3| = 2(2\pi/a)^3$$

$$V_k = (2\pi)^3$$

reciprocal to 
$$\mathbf{b}_1 = (2\pi/a)(\hat{\mathbf{y}} + \hat{\mathbf{z}})$$
;  $\mathbf{b}_2 = (2\pi/a)(\hat{\mathbf{x}} + \hat{\mathbf{z}})$ ;  $\mathbf{b}_3 = (2\pi/a)(\hat{\mathbf{x}} + \hat{\mathbf{y}})$ 

**primitive lattice vectors of an fcc lattice** 

The shortest G's are the following 12 vectors,

$$(2\pi/a)(\pm\hat{\mathbf{y}}\pm\hat{\mathbf{z}})~;~(2\pi/a)(\pm\hat{\mathbf{x}}\pm\hat{\mathbf{z}})~;~(2\pi/a)(\pm\hat{\mathbf{x}}\pm\hat{\mathbf{y}})$$

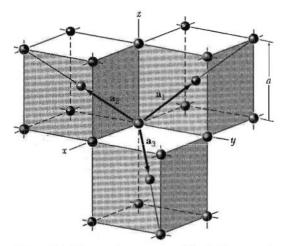


Figure 12 Primitive basis vectors of the body-centered cubic lattice.

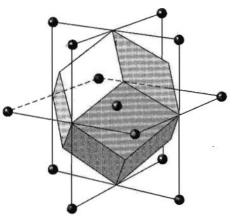


Figure 13 First Brillouin zone of the bodycentered cubic lattice. The figure is a regular rhombic dodecahedron.

1<sup>st</sup> BZ of bcc lattice

#### Reciprocal Lattice to fcc Lattice

The primitive translation vectors of the fcc lattice of Fig. 14 are

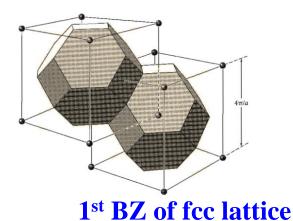
$$\mathbf{a}_1 = \frac{1}{2} a(\hat{\mathbf{y}} + \hat{\mathbf{z}}) \; ; \quad \mathbf{a}_2 = \frac{1}{2} a(\hat{\mathbf{x}} + \hat{\mathbf{z}}) \; ; \quad \mathbf{a}_3 = \frac{1}{2} a(\hat{\mathbf{x}} + \hat{\mathbf{y}}) \; .$$

reciprocal to

$$\mathbf{b}_1 = (2\pi/a)(-\hat{\mathbf{x}} + \hat{\mathbf{y}} + \hat{\mathbf{z}}) ; \qquad \mathbf{b}_2 = (2\pi/a)(\hat{\mathbf{x}} - \hat{\mathbf{y}} + \hat{\mathbf{z}})$$
$$\mathbf{b}_3 = (2\pi/a)(\hat{\mathbf{x}} + \hat{\mathbf{y}} - \hat{\mathbf{z}}) .$$

primitive lattice vectors of a bcc lattice

The shortest **G**'s are the eight vectors:  $(2\pi/a)(\pm \hat{\mathbf{x}} \pm \hat{\mathbf{y}} \pm \hat{\mathbf{z}})$ 



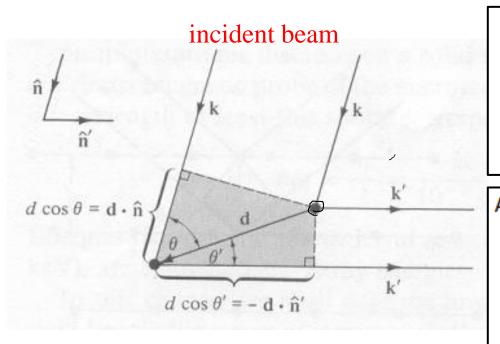
The volume of the primitive cell is

$$V = |\mathbf{a}_1 \cdot \mathbf{a}_2 \times \mathbf{a}_3| = \frac{1}{4} a^3$$

Q. What is the volume of the 1<sup>st</sup> BZ of fcc lattice?

#### Von Laue formulation

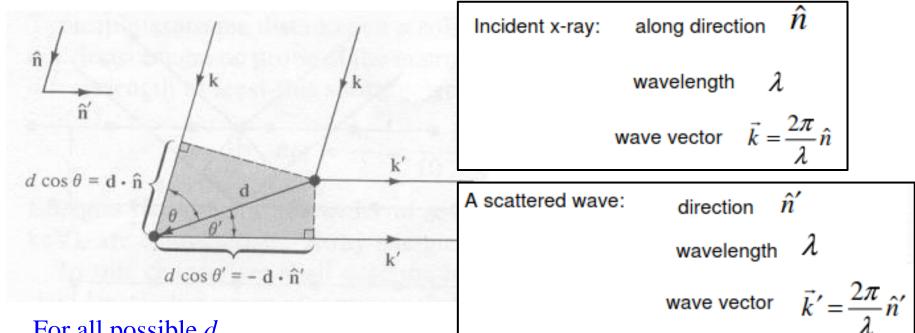
: diffraction condition in terms of scattering by lattice point ( > charge distribution)



Incident x-ray: along direction  $\hat{n}$  wavelength  $\lambda$  wave vector  $\vec{k} = \frac{2\pi}{\lambda} \hat{n}$ 

A scattered wave: direction  $\hat{n}'$  wavelength  $\lambda$  wave vector  $\vec{k}' = \frac{2\pi}{\lambda} \hat{n}'$ 

## Von Laue formulation of X-ray diffraction by a crystal



For all possible d,

$$d\cos\theta + d\cos\theta' = \vec{d}\cdot(\hat{n} - \hat{n}') = m\lambda$$
  $(m:\text{integer})$   $\times (\frac{2\pi}{\lambda})$ 

For all possible **T**,

$$T = u_1 \mathbf{a}_1 + u_2 \mathbf{a}_2 + u_3 \mathbf{a}_3.$$

$$\vec{T} \cdot (\vec{k} - \vec{k}') = 2\pi m \qquad \Delta k = \vec{k}' - \vec{k}$$

$$e^{i(\vec{k}' - \vec{k}) \cdot \vec{T}} = 1$$

→ **Diffraction** condition

## **Reciprocal lattice vectors**

$$\mathbf{b}_1 = 2\pi \frac{\mathbf{a}_2 \times \mathbf{a}_3}{\mathbf{a}_1 \cdot \mathbf{a}_2 \times \mathbf{a}_3} \; ; \qquad \mathbf{b}_2 = 2\pi \frac{\mathbf{a}_3 \times \mathbf{a}_1}{\mathbf{a}_1 \cdot \mathbf{a}_2 \times \mathbf{a}_3} \; ; \qquad \mathbf{b}_3 = 2\pi \frac{\mathbf{a}_1 \times \mathbf{a}_2}{\mathbf{a}_1 \cdot \mathbf{a}_2 \times \mathbf{a}_3} \; .$$

If  $\mathbf{a_i}$  are primitive,  $\mathbf{b_j}$  are primitive.

$$\mathbf{G} = v_1 \mathbf{b}_1 + v_2 \mathbf{b}_2 + v_3 \mathbf{b}_3$$

$$\exp(i\mathbf{G} \cdot \mathbf{T}) = 1$$
Translational vector in reciprocal lattice space

#### **Laue diffraction condition**

$$e^{i(\vec{k}'-\vec{k})\cdot\vec{R}}=1$$

$$\Delta k = G$$

$$G = reciprocal \ lattice \ vector$$

## **Reciprocal lattice vectors**

 $a_i$ : real-space lattice vector

 $b_i$ : reciprocal space lattice vector

$$b_1 = 2\pi \frac{a_2 \times a_3}{a_1 \cdot (a_2 \times a_3)}, \quad b_2 = 2\pi \frac{a_3 \times a_1}{a_1 \cdot (a_2 \times a_3)}, \quad b_3 = 2\pi \frac{a_1 \times a_2}{a_1 \cdot (a_2 \times a_3)}.$$

 $\rightarrow$  If  $\mathbf{a_i}$  are primitive,  $\mathbf{b_i}$  are primitive.

## **Reciprocity condition**

$$\mathbf{b}_i \cdot \mathbf{a}_j = 2\pi \delta_{ij}$$

 $u_i, v_i$ : integers

$$\mathbf{T} = u_1 \mathbf{a}_1 + u_2 \mathbf{a}_2 + u_3 \mathbf{a}_3.$$

Translational vector in <u>real space</u>

$$\mathbf{G} = v_1 \mathbf{b}_1 + v_2 \mathbf{b}_2 + v_3 \mathbf{b}_3$$
$$\exp(i\mathbf{G} \cdot \mathbf{T}) = 1$$

$$\exp(i\mathbf{G}\cdot\mathbf{T}) = 1$$

**G**: Translational vector

in reciprocal lattice space

숙제: Reciprocal lattice to

- (1) sc
- **(2) fcc**
- **(3) bcc**
- **(4) hcp**

#### **SCATTERED WAVE**

## Fourier analysis of the basis

Scattering amplitude for  $\Delta k = G$ 

$$\Delta \mathbf{k} = \mathbf{G}$$

**Structure factor (integral over a cell)** 

$$F_{\mathbf{G}} = N \int_{cell} d\mathbf{V} \, n(\mathbf{r}) \, \exp(-i\mathbf{G} \cdot \mathbf{r}) = N \underline{S_{\mathbf{G}}}$$

$$S_{\mathbf{G}} = n_{\mathbf{G}} = V_{c}^{-1} \int_{cell} d\mathbf{V} \, n(\mathbf{r}) \, \exp(-i\mathbf{G} \cdot \mathbf{r})$$

$$S_G = n_G = V_c^{-1} \int_{\text{cell}} dV \, n(\mathbf{r}) \, \exp(-i\mathbf{G} \cdot \mathbf{r})$$

$$n(\mathbf{r}) = \sum_{j=1}^{s} n_j (\mathbf{r} - \underline{\mathbf{r}_j})$$

$$S_{\mathbf{G}} = \sum_{j} \int dV \, n_{j}(\mathbf{r} - \mathbf{r}_{j}) \, \exp(-i\mathbf{G} \cdot \mathbf{r})$$

$$= \sum_{j} \exp(-i\mathbf{G} \cdot \mathbf{r}_{j}) \int d\mathbf{V} \, n_{j}(\boldsymbol{\rho}) \, \exp(-i\mathbf{G} \cdot \boldsymbol{\rho})$$

$$S_{\mathbf{c}} = \sum_{j} f_{j} \exp(-i\mathbf{G} \cdot \mathbf{r}_{j})$$
 Structure factor of the basis

$$f_{j} = \int dV \, n_{j}(\boldsymbol{\rho}) \, \exp(-i\mathbf{G} \cdot \boldsymbol{\rho})$$
 Atomic form factor

## SCATTERED WAVE (정리)

## Fourier analysis of the basis

 $n(\mathbf{r} + \mathbf{T}) = n(\mathbf{r})$  $n(\mathbf{r}) = \sum_{\mathbf{G}} n_{\mathbf{G}} \exp(i\mathbf{G} \cdot \mathbf{r})$ 

#### **Scattering amplitude**

(integral over a sample)

$$F = \int d\mathbf{V} \, n(\mathbf{r}) \, \exp[i(\mathbf{k} - \mathbf{k}') \cdot \mathbf{r}] = \int d\mathbf{V} \, n(\mathbf{r}) \, \exp(-i\Delta \mathbf{k} \cdot \mathbf{r})$$

$$= 0 \text{ unless } \Delta k = G$$

For 
$$\Delta \mathbf{k} = \mathbf{G}$$
,  $F = N(V_c n_G) = Nn_G = NS_G$ 

## **Scattering amplitude** for

 $\Delta \mathbf{k} = \mathbf{G}$ 

**Structure factor (integral over a cell)** 

$$F_{\mathbf{G}} = N \int_{cell} d\mathbf{V} \, n(\mathbf{r}) \, \exp(-i\mathbf{G} \cdot \mathbf{r}) = N S_{\mathbf{G}}$$

$$S_{\mathbf{G}} = n_{\mathbf{G}} = V_{c}^{-1} \int_{cell} d\mathbf{V} \, n(\mathbf{r}) \, \exp(-i\mathbf{G} \cdot \mathbf{r})$$

$$S_G = n_G = V_c^{-1} \int_{\text{cell}} dV \, n(\mathbf{r}) \, \exp(-i\mathbf{G} \cdot \mathbf{r})$$

 $n_{\mathbf{G}} = V_c^{-1} \int_{\mathbf{cell}} dV \, n(\mathbf{r}) \, \exp(-i\mathbf{G} \cdot \mathbf{r})$ 

$$S_{\mathbf{c}} = \sum_{i} f_{i} \exp(-i\mathbf{G} \cdot \mathbf{r}_{i})$$
 Structure factor of the basis

$$f_j = \int d\mathbf{V} \, n_j(\mathbf{\rho}) \, \exp(-i\mathbf{G} \cdot \mathbf{\rho})$$

 $f_i = \int dV \, n_i(\rho) \exp(-i\mathbf{G} \cdot \rho)$  Atomic form factor (for each basis atom)

#### **SCATTERED WAVE**

## Classical calculation of the scattering factor

$$f_j = \int d\mathbf{V} \, n_j(\mathbf{\rho}) \, \exp(-i\mathbf{G} \cdot \mathbf{\rho})$$

 $f_j = \int dV \, n_j(\rho) \exp(-i\mathbf{G} \cdot \rho)$  Atomic form factor (for each basis atom)

#### For spherically symmetric charge distribution

$$f_{j} = 2\pi \int dr \, r^{2} \, d(\cos \alpha) \, n_{j}(r) \exp(-iGr \cos \alpha)$$

$$= 2\pi \int dr \, r^{2} n_{j}(r) \left( \frac{e^{iGr} - e^{-iGr}}{iGr} \right), \qquad \frac{2 \sin Gr}{Gr}$$

(i) If 
$$n_j(\mathbf{r}) \sim \delta(0)$$
  $f_j = 4\pi \int d\mathbf{r} \, n_j(\mathbf{r}) r^2 = Z$ 

(ii) Forward scattering G = 0,  $f_i = Z$ 

The overall electron distribution in a solid as seen in x-ray diffraction is fairly close to that of the appropriate free atoms. This statement does not mean that the outermost or valence electrons are not redistributed somewhat in forming the solid; it means only that the x-ray reflection intensities are represented well by the free atom values of the form factors and are not very sensitive to small redistributions of the electrons.

## **SCATTERED WAVE**

## Fourier analysis of the basis

#### **Homework: Prove**

$$\begin{split} S_{\mathbf{G}}(v_1v_2v_3) &= \sum_j f_j \exp[-i2\pi(v_1x_j + v_2y_j + v_3z_j)] \\ &\qquad \qquad \text{with} \qquad \mathbf{r}_j = x_j\mathbf{a}_1 + y_j\mathbf{a}_2 + z_j\mathbf{a}_3 \,, \qquad \mathbf{G} = v_1\mathbf{b}_1 + v_2\mathbf{b}_2 + v_3\mathbf{b}_3 \end{split}$$

#### Structure Factor of the bcc Lattice

The bcc basis referred to the cubic cell has identical atoms at  $x_1 = y_1 = z_1 = 0$  and at  $x_2 = y_2 = z_2 = \frac{1}{2}$ . Thus (46) becomes

bcc lattice = 2 identical basis atoms in a sc cell

a sc reciprocal lattice (diffraction points)

(diffraction intensity)  $\propto$  (the structure factor)<sup>2</sup>  $\left[S_{\mathbf{G}} = \sum_{j} f_{j} \exp(-i\mathbf{G} \cdot \mathbf{r}_{j})\right]^{2}$ 

$$\begin{split} S_{\mathbf{G}}(v_1v_2v_3) &= \sum_j f_j \exp[-i2\pi(v_1x_j + v_2y_j + v_3z_j)] & r_1 &= (0,0,0) \\ & r_2 &= (1/2,1/2,1/2) \\ &= f[1 + \exp[-i\pi(v_1 + v_2 + v_3)]] \end{split}$$

$$S = 0$$
 when  $v_1 + v_2 + v_3 = \text{odd integer}$ ;  
 $S = 2f$  when  $v_1 + v_2 + v_3 = \text{even integer}$ .



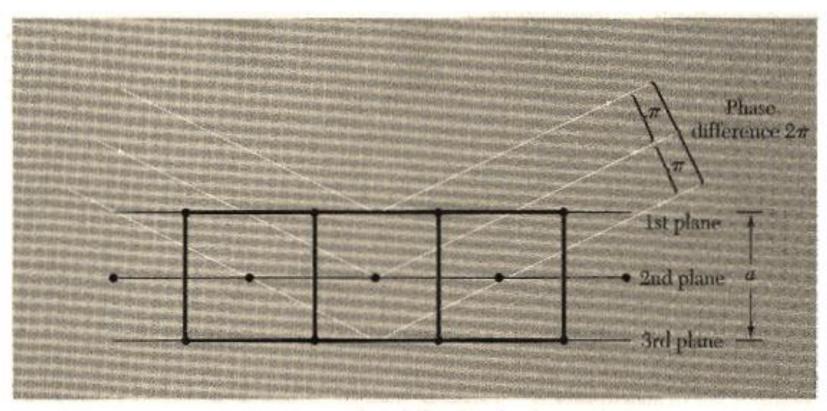


Figure 16 Explanation of the absence of a (100) reflection from a body-centered cubic lattice. The phase difference between successive planes is  $\pi$ , so that the reflected amplitude from two adjacent planes is  $1 + e^{-i\pi} = 1 - 1 = 0$ .

#### Structure Factor of the fcc Lattice

The basis of the fcc structure referred to the cubic cell has identical atoms at 000;  $0^{11}_{22}$ ;  $\frac{1}{2}0^{1}_{2}$ ;  $\frac{11}{22}0$ . Thus (46) becomes

$$fcc$$
 lattice = 4 identical basis atoms in a sc cell

a sc reciprocal lattice (diffraction points)

$$r_1 = (0,0,0)$$
  
 $r_2 = (0,1/2,1/2)$   
 $r_3 = (1/2,0,1/2)$   
 $r_4 = (1/2,1/2,0)$ 

$$S(v_1v_2v_3) = f[1 + \exp[-i\pi(v_2 + v_3)] + \exp[-i\pi(v_1 + v_3)] + \exp[-i\pi(v_1 + v_2)]]$$

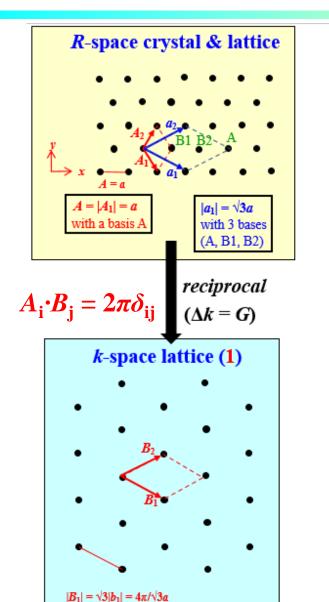
$$+ \exp[-i\pi(v_1 + v_2)] .$$
(48)

If all indices are even integers, S=4f; similarly if all indices are odd integers. But if only one of the integers is even, two of the exponents will be odd multiples of  $-i\pi$  and S will vanish. If only one of the integers is odd, the same argument applies and S will also vanish. Thus in the fcc lattice no reflections can occur for which the indices are partly even and partly odd.



Non-zero intensities result in **bcc** diffraction points (reciprocal lattice)

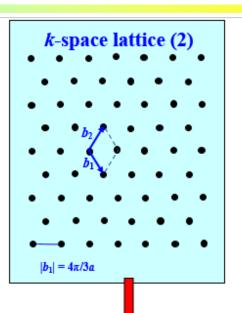
## 2D example



$$a_{i} \cdot b_{j} = 2\pi \delta_{ij}$$

$$reciprocal$$

$$(\Delta k = G)$$



 $(f_{\mathcal{A}} = f_{B1} = f_{B2})$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$   $\bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet \quad \bullet$ 

 $|B_1| = \sqrt{3}|b_1| = 4\pi/\sqrt{3}a$ 

Considering structure factor  $S_G$ 

- For a given crystal, there are many way to take "lattice & basis".
- For different choices of "lattice & basis", the reciprocal lattices are different.
- But, the diffraction pattern and intensity from the crystal should be unique because some reciprocal lattice poits become extinct when the structure factor (diffraction intensity) is taken into account.

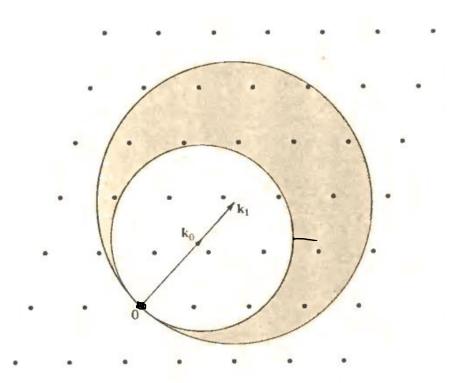
(diffraction intensity)  $\propto$  (the structure factor)<sup>2</sup>

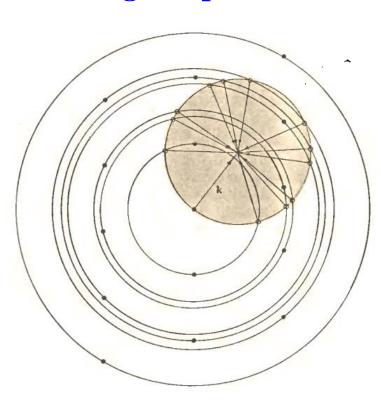
## How to do diffraction experiment (XRD)

 $\Delta \mathbf{k} = \mathbf{G}$ 

#### Laue method

## **Rotating sample method**

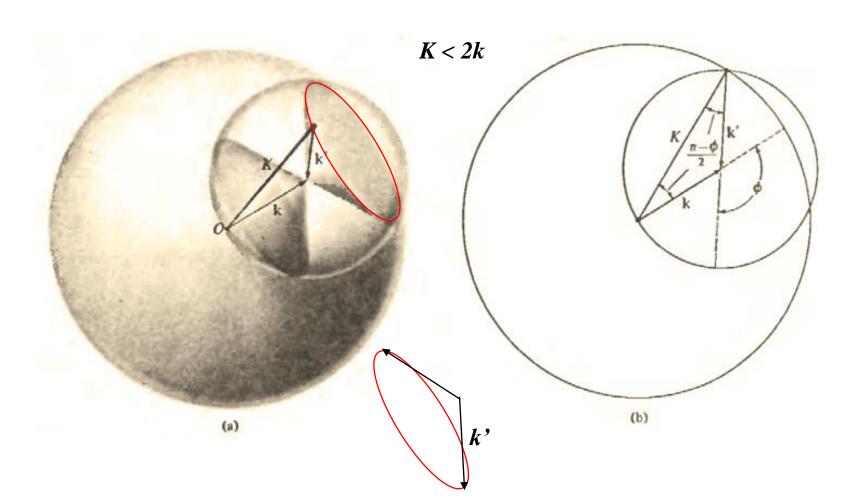




## How to do diffraction experiment (XRD)

$$\Delta \mathbf{k} = \mathbf{G}$$

#### **Powder method**



#### **SUMMARY**

#### SUMMARY

• Various statements of the Bragg condition:

$$2d \sin \theta = n\lambda \; ; \qquad \Delta \mathbf{k} = \mathbf{G} \; ; \qquad 2\mathbf{k} \cdot \mathbf{G} = \mathbf{G}^2 \; .$$

Laue conditions:

$$\mathbf{a}_1 \cdot \Delta \mathbf{k} = 2\pi v_1$$
;  $\mathbf{a}_2 \cdot \Delta \mathbf{k} = 2\pi v_2$ ;  $\mathbf{a}_3 \cdot \Delta \mathbf{k} = 2\pi v_3$ .

· The primitive translation vectors of the reciprocal lattice are

$$\mathbf{b}_1 = 2\pi \frac{\mathbf{a}_2 \times \mathbf{a}_3}{\mathbf{a}_1 \cdot \mathbf{a}_2 \times \mathbf{a}_3} \; ; \qquad \mathbf{b}_2 = 2\pi \frac{\mathbf{a}_3 \times \mathbf{a}_1}{\mathbf{a}_1 \cdot \mathbf{a}_2 \times \mathbf{a}_3} \; ; \qquad \mathbf{b}_3 = 2\pi \frac{\mathbf{a}_1 \times \mathbf{a}_2}{\mathbf{a}_1 \cdot \mathbf{a}_2 \times \mathbf{a}_3} \; .$$

Here  $a_1$ ,  $a_2$ ,  $a_3$  are the primitive translation vectors of the crystal lattice.

A reciprocal lattice vector has the form

$$\mathbf{G} = v_1 \mathbf{b}_1 + v_2 \mathbf{b}_2 + v_3 \mathbf{b}_3 ,$$

where  $v_1, v_2, v_3$  are integers or zero.

• The scattered amplitude in the direction  $\mathbf{k}' = \mathbf{k} + \Delta \mathbf{k} = \mathbf{k} + \mathbf{G}$  is proportional to the geometrical structure factor:

$$S_{\mathbf{G}} = \sum f_j \exp(-i\mathbf{r}_j \cdot \mathbf{G}) = \sum f_j \exp[-i2\pi(x_j v_1 + y_j v_2 + z_j v_3)] ,$$

where j runs over the s atoms of the basis, and  $f_j$  is the atomic form factor (49) of the jth atom of the basis. The expression on the right-hand side is written for a reflection  $(v_1v_2v_3)$ , for which  $\mathbf{G} = v_1\mathbf{b}_1 + v_2\mathbf{b}_2 + v_3\mathbf{b}_3$ .

 Any function invariant under a lattice translation T may be expanded in a Fourier series of the form

$$n(\mathbf{r}) = \sum_{\mathbf{G}} n_{\mathbf{G}} \exp(i\mathbf{G} \cdot \mathbf{r})$$
.

 The first Brillouin zone is the Wigner-Seitz primitive cell of the reciprocal lattice. Only waves whose wavevector k drawn from the origin terminates on a surface of the Brillouin zone can be diffracted by the crystal.

Simple cubic Cube

Body-centered cubic Rhombic dodecahedron (Fig. 13) Face-centered cubic Truncated octahedron (Fig. 15)

## Homework