

A Variational Framework for Phase-Field Fracture Modeling with Applications to Fragmentation, Desiccation, Ductile Failure, and Spallation

Dissertation Defense

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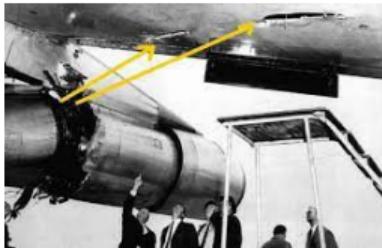
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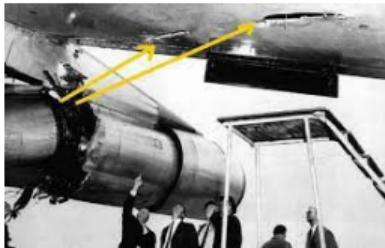
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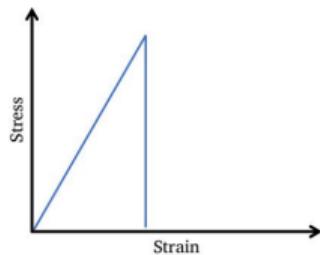


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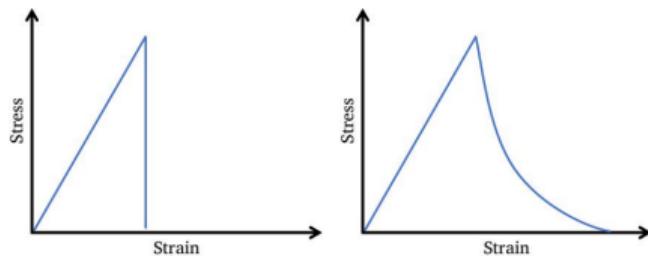


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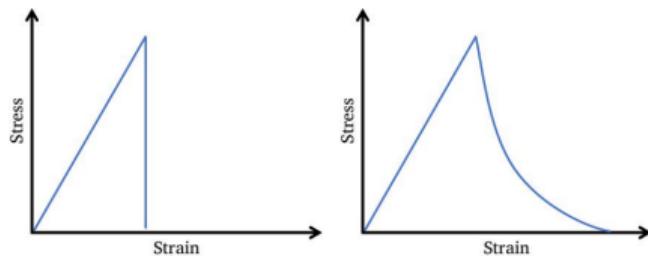
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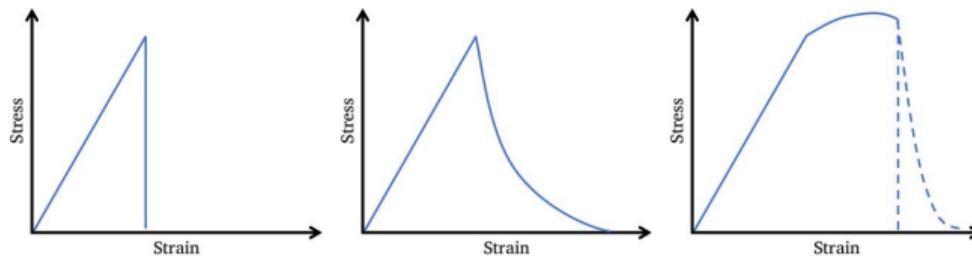
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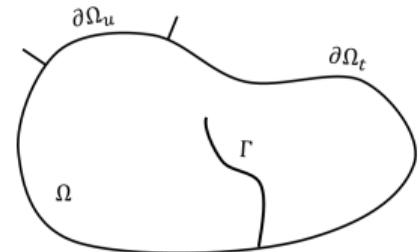
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To date, fracture is still one of the most challenging phenomena to model and predict.

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is approximated with

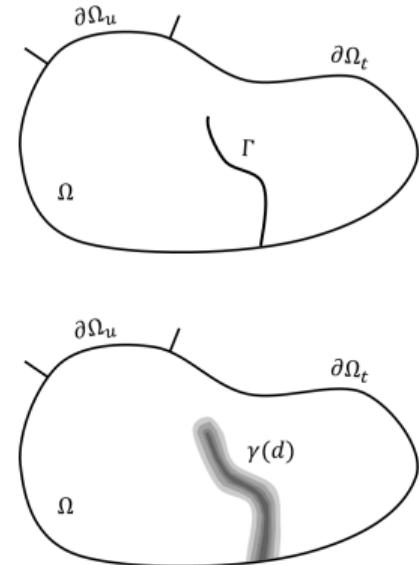
the crack surface density function $\gamma = \hat{\gamma}_l(d)$:

$$\Psi^f \approx \int_{\Omega} \mathcal{G}_c \gamma \, dV, \quad \gamma = \frac{1}{c_0 l} \left(\alpha + l^2 \nabla d \cdot \nabla d \right).$$

- $d \in [0, 1]$ is the phase field;
- $\alpha = \hat{\alpha}(d)$ is the crack geometric function, $\hat{\alpha}(0) = 0$, $\hat{\alpha}(1) = 1$;
- $g = \hat{g}(d)$ is the degradation function, $\hat{g}(0) = 1$, $\hat{g}(1) = 0$;
- c_0 is chosen such that

$$\lim_{l \rightarrow 0^+} \int_{\Omega} \mathcal{G}_c \gamma \, dV = \int_{\Gamma} \mathcal{G}_c \, dA.$$

See [1] for more details.



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$$\Phi, \quad \mathbf{F}^p, \quad \bar{\varepsilon}^p, \quad d, \quad T.$$

- Conservations and thermodynamic laws:

$$\dot{\rho}_0 = 0,$$

$$\rho_0 \mathbf{a} = \nabla \cdot \mathbf{P} + \rho_0 \mathbf{b},$$

$$\mathbf{P}\mathbf{F} = \mathbf{F}\mathbf{P}^T,$$

$$f - \nabla \cdot \boldsymbol{\xi} = 0,$$

$$\dot{u} + \dot{k} = \mathcal{P}^{\text{ext}} + \rho_0 q - \nabla \cdot \mathbf{h},$$

$$\dot{s}^{\text{int}} = \dot{s} - \frac{\rho_0 q}{T} + \nabla \cdot \frac{\mathbf{h}}{T} \geq 0.$$

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- Generalized forces are

$$\begin{aligned} \mathbf{P} &= \mathbf{P}^{\text{eq}} + \mathbf{P}^{\text{vis}}, \quad \mathbf{T} = \mathbf{T}^{\text{eq}} + \mathbf{T}^{\text{vis}}, \quad Y = Y^{\text{eq}} + Y^{\text{vis}}, \\ f &= f^{\text{eq}} + f^{\text{vis}}, \quad \boldsymbol{\xi} = \boldsymbol{\xi}^{\text{eq}} + \boldsymbol{\xi}^{\text{vis}}, \end{aligned}$$

- Following the Coleman-Noll procedure:

$$\begin{aligned} \mathbf{P}^{\text{eq}} &= \psi, \mathbf{F}, \quad \mathbf{T}^{\text{eq}} = \psi, \mathbf{F}^p, \quad Y^{\text{eq}} = \psi, \bar{\varepsilon}^p, \\ f^{\text{eq}} &= \psi, d, \quad \boldsymbol{\xi}^{\text{eq}} = \psi, \nabla d, \quad -s = \psi, T. \end{aligned}$$

- Viscous forces follow from the dual kinetic potential Δ^* :

$$\begin{aligned} \mathbf{P}^{\text{vis}} &= \Delta^*, \quad \mathbf{T}^{\text{vis}} = \Delta^*, \quad Y^{\text{vis}} = \Delta^*, \\ f^{\text{vis}} &= \Delta^*, \quad \boldsymbol{\xi}^{\text{vis}} = \Delta^*, \end{aligned}$$

- To satisfy the second law:

$$\delta = \mathbf{P}^{\text{vis}} : \dot{\mathbf{F}} + \mathbf{T}^{\text{vis}} : \dot{\mathbf{F}}^p + Y^{\text{vis}} \dot{\bar{\varepsilon}}^p + f^{\text{vis}} \dot{d} + \boldsymbol{\xi}^{\text{vis}} \cdot \nabla d \geq 0.$$

With $\mathcal{V} = \{\dot{\phi}, \dot{\mathbf{F}}^p, \dot{\bar{\varepsilon}}^p, \dot{d}\}$:

$$(\mathcal{V}, \dot{s}, T) = \arg \left(\inf_{\mathcal{V}, \dot{s}} \sup_T L \right)$$

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Benefits:

- From a theoretical standpoint:
 - The direct method of calculus of variations informs conditions for the existence and uniqueness of solutions.
 - Localization effects can be studied within the framework of free-discontinuity problems.
- From a computational standpoint:
 - Discretization leads to a symmetric operator.
 - Discretization leads to robust and efficient variational constitutive update.
 - The total potential can assist line search.
 - The total potential can be directly used as an error indicator for adaptive mesh refinement.
 - Many powerful optimization packages exist, e.g. PETSc/TAO, Trilinos, Matlab, etc..

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- Construction of such a potential is no easy task.
- (Other limitations will be discussed in the end.)

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State-of-the-art: variational brittle fracture that concerns with

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while we are looking at

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The total potential L is constructed as

$$L = \int_{\Omega_0} \varphi \, dV - \mathcal{P}^{\text{ext}},$$

$$\varphi = \dot{k} + \dot{u} + \Delta^* - T\dot{s} - \chi,$$

The external power expenditure $\mathcal{P}^{\text{ext}}(\dot{\phi}, T)$ is defined as

$$\begin{aligned} \mathcal{P}^{\text{ext}} = & \underbrace{\int_{\Omega_0} \rho_0 \mathbf{b} \cdot \dot{\phi} \, dV}_{\text{body force}} + \underbrace{\int_{\partial_t \Omega_0} \mathbf{t} \cdot \dot{\phi} \, dA}_{\text{surface traction}} + \underbrace{\int_{\partial_h \Omega_0} \bar{h}_n \ln\left(\frac{T}{T_0}\right) \, dA}_{\text{external heat flux}} \\ & + \underbrace{\int_{\partial_r \Omega_0} h \left[T - T_0 \ln\left(\frac{T}{T_0}\right) \right] \, dA}_{\text{external heat convection}} - \underbrace{\int_{\Omega_0} \rho_0 q \ln\left(\frac{T}{T_0}\right) \, dV}_{\text{heat source}}, \end{aligned}$$

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- (Strict) dissipation inequality requires Δ^* to be convex in each rate.

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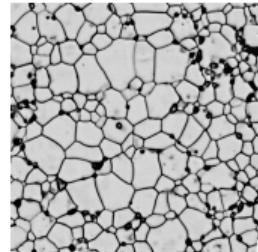
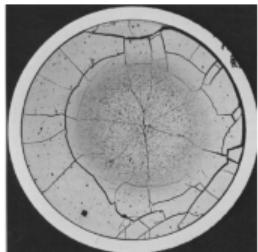
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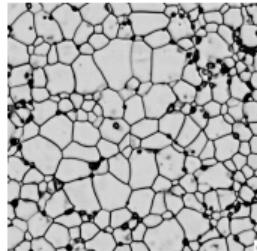
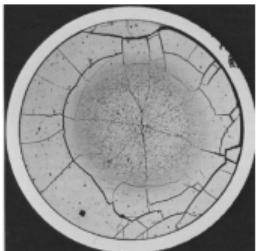
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Background:

- Fission of UO₂ produces a variety of fission products.
- Properties of UO₂ are strongly influenced by fracture.
- Gas bubbles, grains, and grain boundaries alter fracture properties.
- Existing 2D models over-simplifies the microstructure and results in inaccurate strength-porosity relations.



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Model:

- Helmholtz free energy density:

$$\psi = \psi^e + \psi^f.$$

- Strain energy density:

$$\psi^e = g\psi_{\langle A \rangle}^e + \psi_{\langle I \rangle}^e,$$

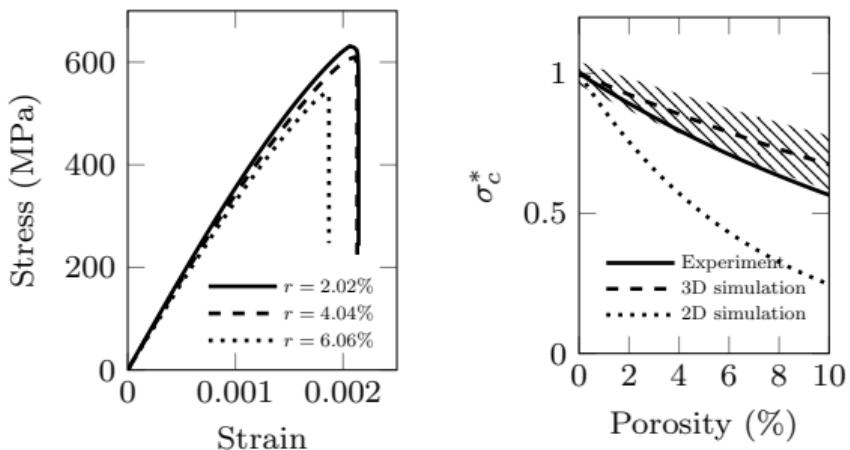
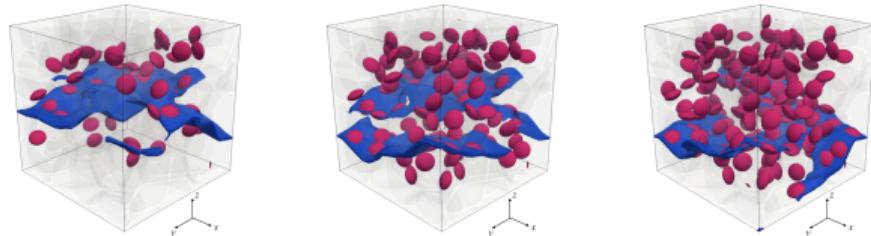
$$\psi_{\langle A \rangle}^e = \frac{1}{2}\lambda \langle \text{tr}(\boldsymbol{\varepsilon}) \rangle_+^2 + G\boldsymbol{\varepsilon}^+ : \boldsymbol{\varepsilon}^+,$$

$$\psi_{\langle I \rangle}^e = \frac{1}{2}\lambda \langle \text{tr}(\boldsymbol{\varepsilon}) \rangle_-^2 + G\boldsymbol{\varepsilon}^- : \boldsymbol{\varepsilon}^-,$$

- Fracture energy density:

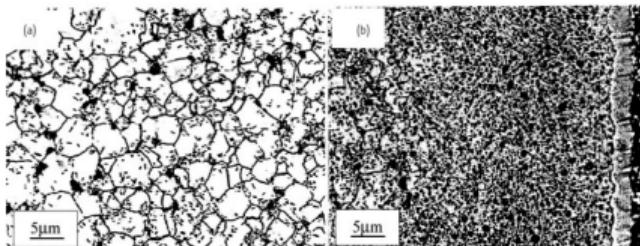
$$\psi^f = \frac{\mathcal{G}_c}{c_0 l} (\alpha + l^2 \nabla d \cdot \nabla d),$$

$$\alpha = d^2, \quad g = (1 - d)^2.$$



- A set of random close-packing voronoi structures are realized by maximal Poisson-disk sampling.
- The microstructure is generated using a phase-field grain growth model [2].
- Grain boundaries have an arbitrarily high fracture toughness to facilitate intergranular fracture.
- Numerical studies are performed to investigate the effects of bubble geometry, loading conditions, and porosity on the critical fracture strength.
- Results of 15 realizations of 3 porosity levels are fitted using the relation suggested by experiments [3]:

$$\frac{\sigma_c}{\sigma_0} = \exp(-ar).$$

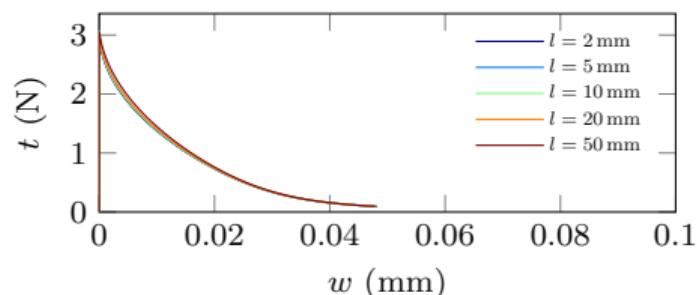
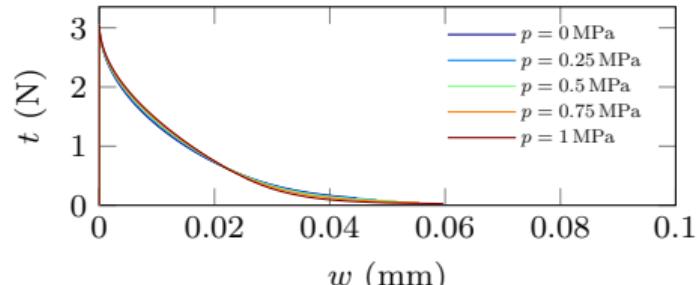


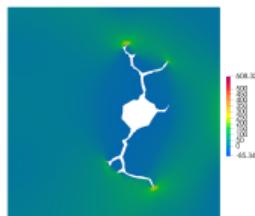
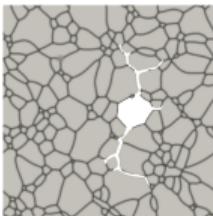
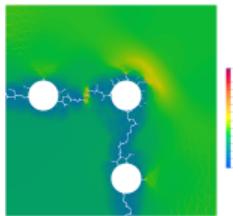
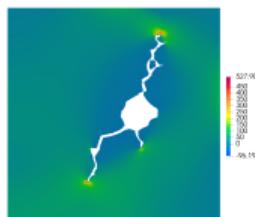
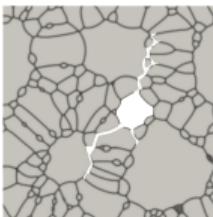
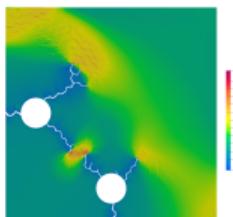
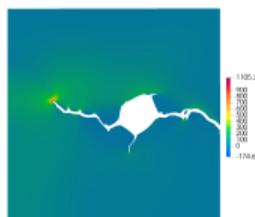
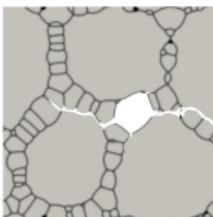
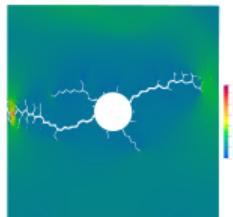
Background:

- Utilities are seeking to increase the allowable burnup limit for UO₂ fuel.
- The risk of fragmentation during a loss-of-coolant accident (LOCA) is a major limitation.
- Over-pressurization of fission gas bubbles results in fine fragmentation of high burnup structures.
- A model for predicting the onset of fragmentation is essential.

External pressure power:

$$\mathcal{P}^{\text{ext}} = - \int_{\Omega_0} \bar{p} \nabla d \cdot \dot{\phi} I_{d,d} \, dV.$$





- A 2D REV is considered. Plane strain conditions are assumed to hold.
- LOCA pressure transients:
 - The temperature as a function of time at the edge of a representative pellet for each rod is obtained from simulations.
 - The temperature transient is used as an input to a Kim-Kim-Suzuki (KKS) phase-field model [4] to determine the pressure transient.
 - The pressure transient is treated as a known in the fracture model.
- Effects of bubble size, bubble pressure, surrounding pressure, and multi-bubble interaction are investigated.
- Defect evolution and recrystallization can be incorporated into the fracture model.

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Background:

- Film-substrate systems widely exist in nature and in engineering applications.
- Fracture of thin films has been studied using model-based simulations based on a wide range of methodologies.
- Many soil materials are “cohesive” in nature. It calls for a phase-field model for cohesive fracture.
- Many film-substrate systems are symmetric or axisymmetric. It is important to incorporate stochastic models for material properties.



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Model:

- Enforcing traction-free boundary conditions:

$$\psi^e = g\psi_{\langle A \rangle}^e + \psi_{\langle I \rangle}^e,$$

$$\psi_{\langle A \rangle}^e = \frac{1}{2}\boldsymbol{\sigma}_{\langle A \rangle} : \boldsymbol{\varepsilon}, \quad \psi_{\langle I \rangle}^e = \frac{1}{2}\boldsymbol{\sigma}_{\langle I \rangle} : \boldsymbol{\varepsilon},$$

$$\boldsymbol{\sigma}_{\langle A \rangle} = \boldsymbol{\sigma}_n^+ + \boldsymbol{\sigma}_t, \quad \boldsymbol{\sigma}_{\langle I \rangle} = \boldsymbol{\sigma}_n^-,$$

$$\boldsymbol{\sigma}_n^\pm = \langle -t_N \rangle_\pm \tilde{\mathbf{n}} \otimes \tilde{\mathbf{n}}, \quad \boldsymbol{\sigma}_t = \boldsymbol{\sigma} - \boldsymbol{\sigma}_n^+ - \boldsymbol{\sigma}_n^-.$$

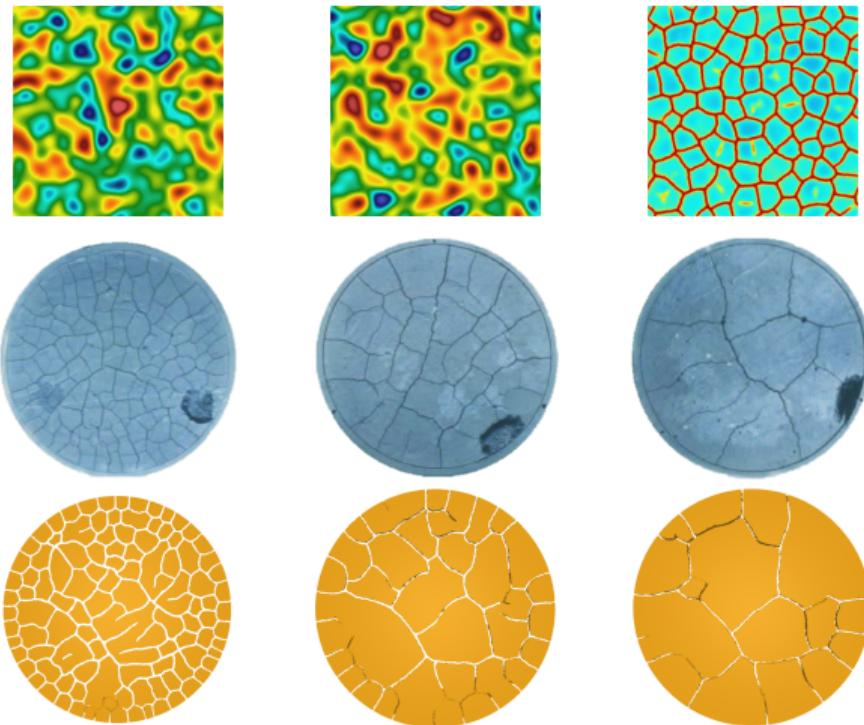
- Cohesive fracture:

$$\alpha = \xi d - (1 - \xi)d^2, \quad g = \frac{1}{1 + \phi},$$

$$\phi = \frac{a_1 d + a_1 a_2 d^2 + a_1 a_2 a_3 d^3}{(1 - d)^p}.$$

- A nonlinear softening law:

$$\xi = 1, \quad p = 2, \quad a_1 = \frac{\mathcal{G}_c}{c_0 l \psi_c}, \quad a_2 = 1, \quad a_3 = 0.$$



- Only “channeling” cracks in the thin film are considered.
- Thermal effects are neglected. Dehydration is modeled as pre-stress (or equivalent eigenstrains).
- The fracture model is verified with analytical solutions in a periodic quasi-1D context.
- Pervasive fracture is studied with a 2D simplification.
- Material property inhomogeneity is represented by two pointwise correlated random fields $\{\mathcal{G}_c(\mathbf{X}), \mathbf{X} \in \Omega\}$ and $\{\psi_c(\mathbf{X}), \mathbf{X} \in \Omega\}$.
- The versatility offered by the probabilistic framework is highlighted by solving a 3D problem based on physical experiments.

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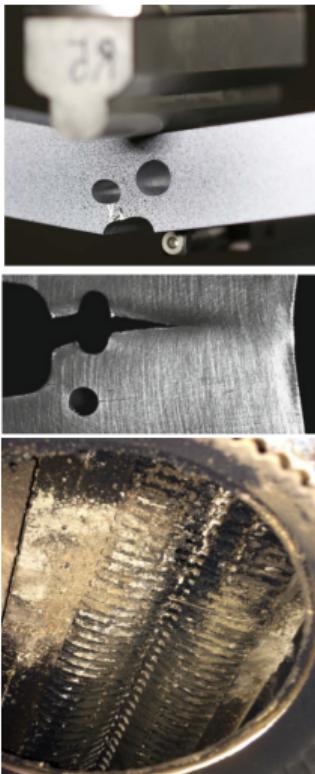
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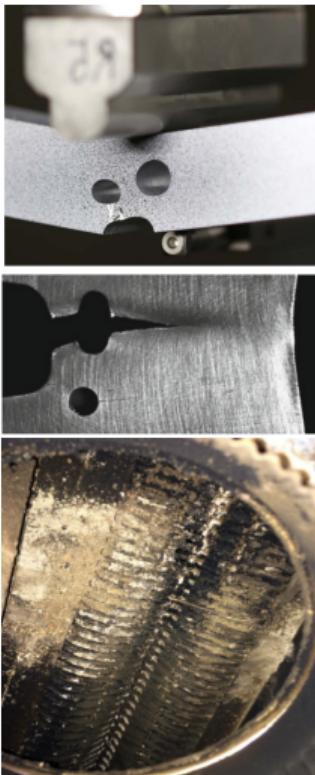
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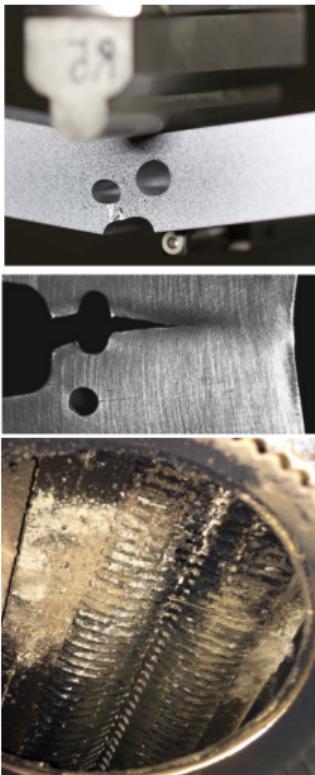
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How to best design the coupling between plasticity and fracture remains an open question. Some challenges/issues:

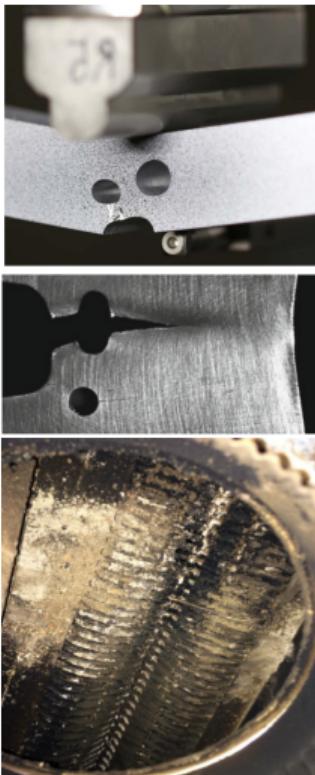
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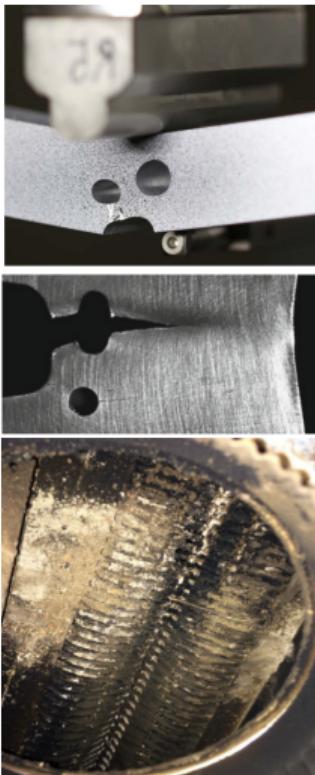
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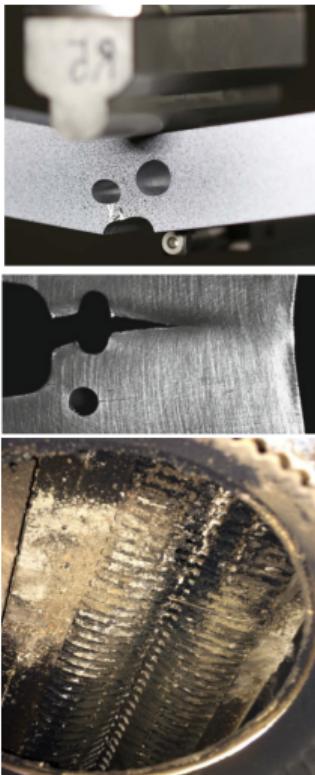
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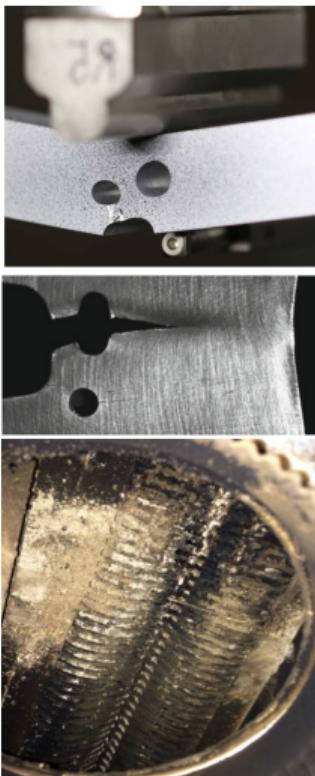
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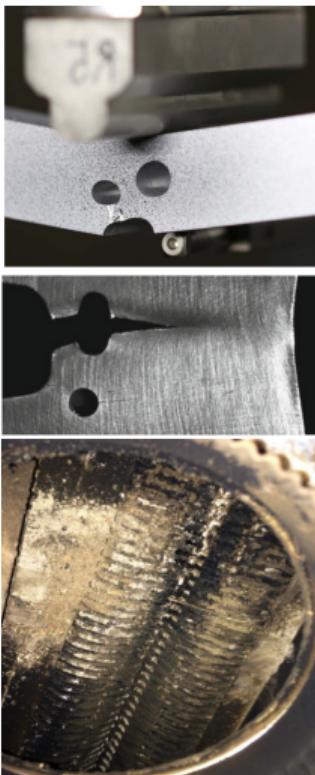
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- How much plastic dissipation contributes to fracture evolution?
- How much plastic dissipation converts to heat generation?

All of the issues above can be addressed by the proposed framework
(with proper constitutive choices).

$$L = \int_{\Omega_0} \left[\dot{\psi}^e + \psi^{e*} + (1 - \mathcal{Q})\dot{\psi}^p + \mathcal{Q}\psi^{p*} + \dot{\psi}^f + \psi^{f*} - T\dot{s} - \chi \right] dV - \mathcal{P}^{\text{ext}},$$

subject to $\mathbf{L}(\mathbf{Z})\dot{\mathbf{Z}} = \mathbf{0}$.

Option 1 (Compressible Neo-Hookean):

$$\begin{aligned} \psi^e &= g^e \psi_{\langle A \rangle}^e + \psi_{\langle I \rangle}^e, \\ \psi_{\langle A \rangle}^e &= \mathbb{H}_1(J) \left\{ \frac{1}{2} K \left[\frac{1}{2} (J^2 - 1) - \ln(J) \right] \right\} \\ &\quad + \frac{1}{2} G (\bar{\mathbf{C}} : \mathbf{C}^{p-1} - 3), \\ \psi_{\langle I \rangle}^e &= (1 - \mathbb{H}_1(J)) \left\{ \frac{1}{2} K \left[\frac{1}{2} (J^2 - 1) - \ln(J) \right] \right\}. \end{aligned}$$

Option 2 (Hencky):

$$\begin{aligned} \psi^e &= g^e \psi_{\langle A \rangle}^e + \psi_{\langle I \rangle}^e, \\ \psi_{\langle A \rangle}^e &= \frac{1}{2} K \langle \text{tr}(\boldsymbol{\varepsilon}^e) \rangle_+^2 + G \text{dev} \boldsymbol{\varepsilon}^e : \text{dev} \boldsymbol{\varepsilon}^e, \\ \psi_{\langle I \rangle}^e &= \frac{1}{2} K \langle \text{tr}(\boldsymbol{\varepsilon}^e) \rangle_-^2, \\ \boldsymbol{\varepsilon}^e &= \frac{1}{2} \ln(\mathbf{C}^e). \end{aligned}$$

$$L = \int_{\Omega_0} \left[\dot{\psi}^e + \psi^{e*} + (1 - \mathcal{Q})\dot{\psi}^p + \mathcal{Q}\psi^{p*} + \dot{\psi}^f + \psi^{f*} - T\dot{s} - \chi \right] dV - \mathcal{P}^{\text{ext}},$$

subject to $\mathbf{L}(\mathbf{Z})\dot{\mathbf{Z}} = \mathbf{0}$.

Option 1 (Linear hardening):

$$\begin{aligned} \psi^e &= g^e \psi_{\langle A \rangle}^e + \psi_{\langle I \rangle}^e, \\ \psi_{\langle A \rangle}^e &= \mathbb{H}_1(J) \left\{ \frac{1}{2} K \left[\frac{1}{2} (J^2 - 1) - \ln(J) \right] \right\} \\ &\quad + \frac{1}{2} G (\bar{\mathbf{C}} : \mathbf{C}^{p-1} - 3), \\ \psi_{\langle I \rangle}^e &= (1 - \mathbb{H}_1(J)) \left\{ \frac{1}{2} K \left[\frac{1}{2} (J^2 - 1) - \ln(J) \right] \right\}. \end{aligned}$$

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