



# Field Limitation in Nb<sub>3</sub>Sn Cavities

Ryan Porter  
Cornell University

Supported by:

U.S. DOE award DE-SC0008431: 1.3 GHz coatings + tests

NSF Award 1734189: 2.6 GHz + 3.9 GHz coatings + tests

Center for Bright Beams , NSF Award PHY-1549132: material studies

This work make use of Cornell Center for Materials Research, NSF MRSEC program (DMR-1719875)



*Center for*  
**BRIGHT BEAMS**  
A National Science Foundation Science & Technology Center





- Introduce Nb<sub>3</sub>Sn
- Standard Nb<sub>3</sub>Sn cavity performance
- Experimental data
- Quench models
- Reducing roughness
- Conclusion

# Properties of Nb<sub>3</sub>Sn

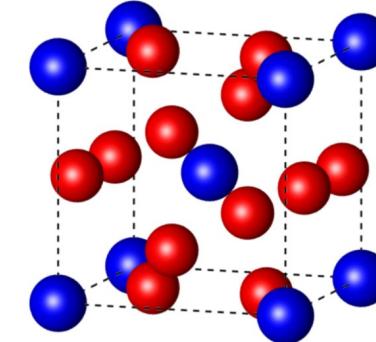
## Higher critical temperature

→ Operation at 4.2 K

## Higher superheating field

→ Double the limit of niobium

Parameter	Niobium	Nb <sub>3</sub> Sn
Transition temperature	9.2 K	18 K
Superheating field	219 mT	425 mT
Energy gap $\Delta/k_b T_c$	1.8	2.2
$\lambda$ at T = 0 K	50 nm	111 nm
$\xi$ at T = 0 K	22 nm	4.2 nm
GL parameter $\kappa$	2.3	26



Blue: tin

Red: niobium

1. Lower losses
2. Higher gradients  
~90 MV/m

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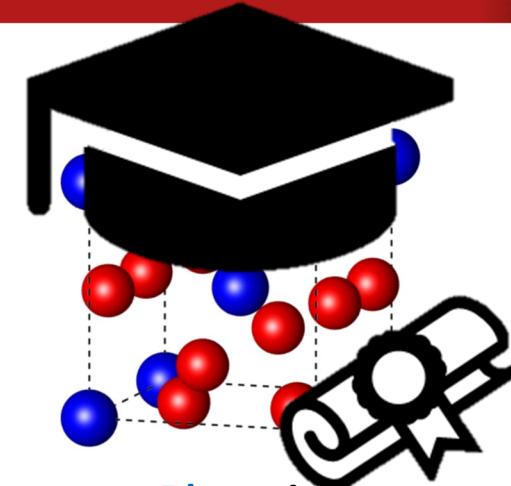
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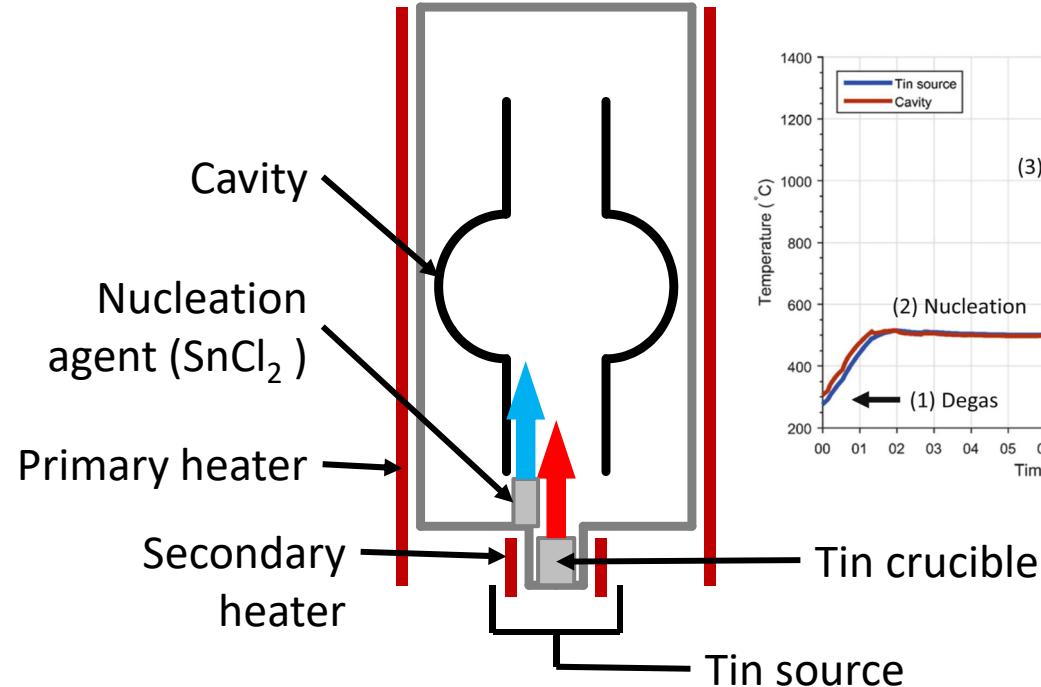


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# Cornell Nb<sub>3</sub>Sn Vapor Diffusion Furnace

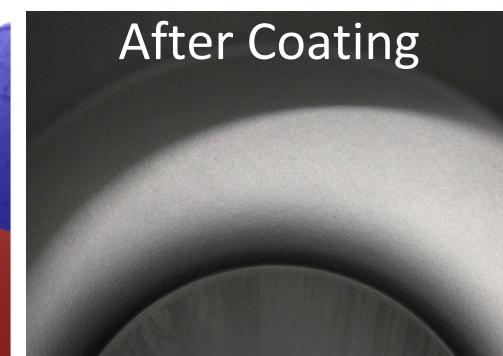
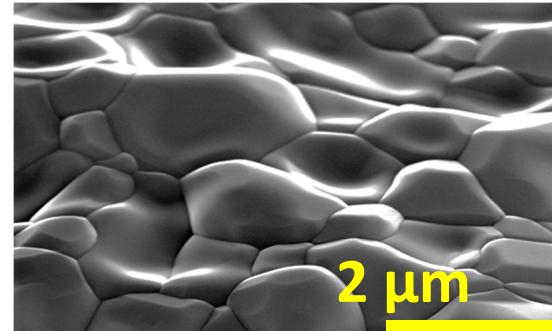
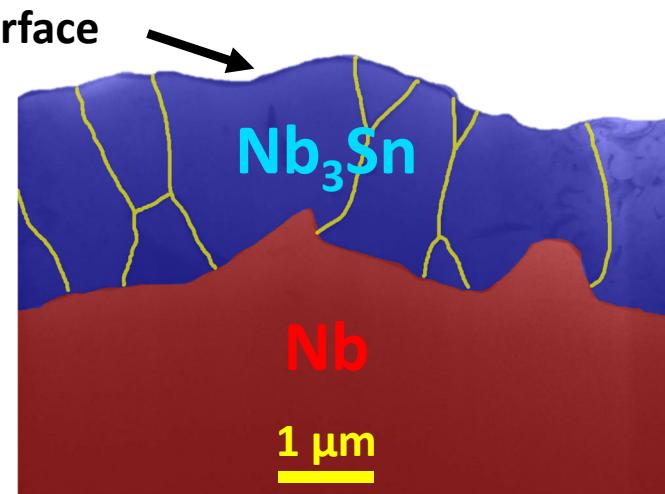
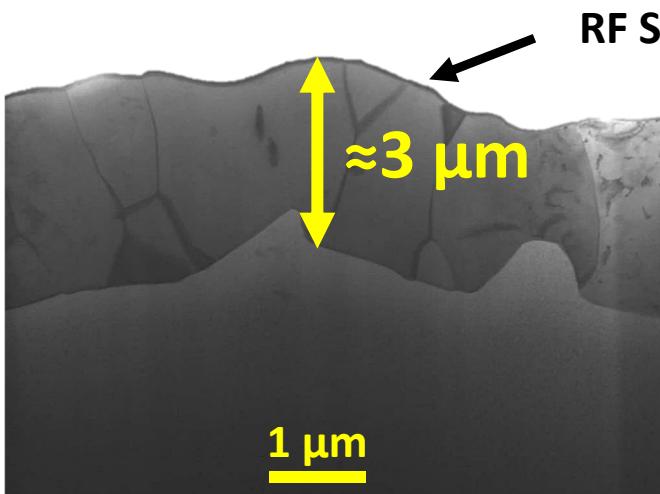


"Wuppertal" configuration, i.e., with secondary heater for the tin source  
Optimized nucleation and temperature profile

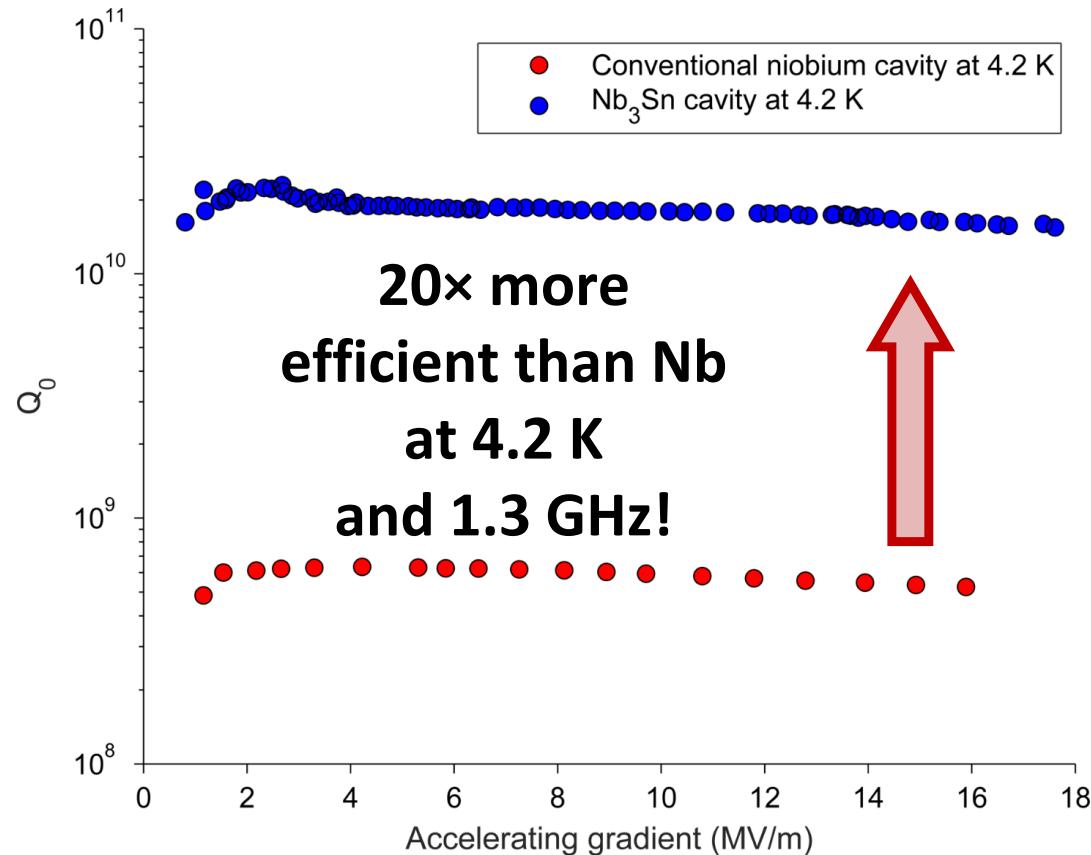
S. Posen and M. Liepe, Phys. Rev. ST Accel. Beams 15, 112001 (2014).

# Nb<sub>3</sub>Sn Coatings

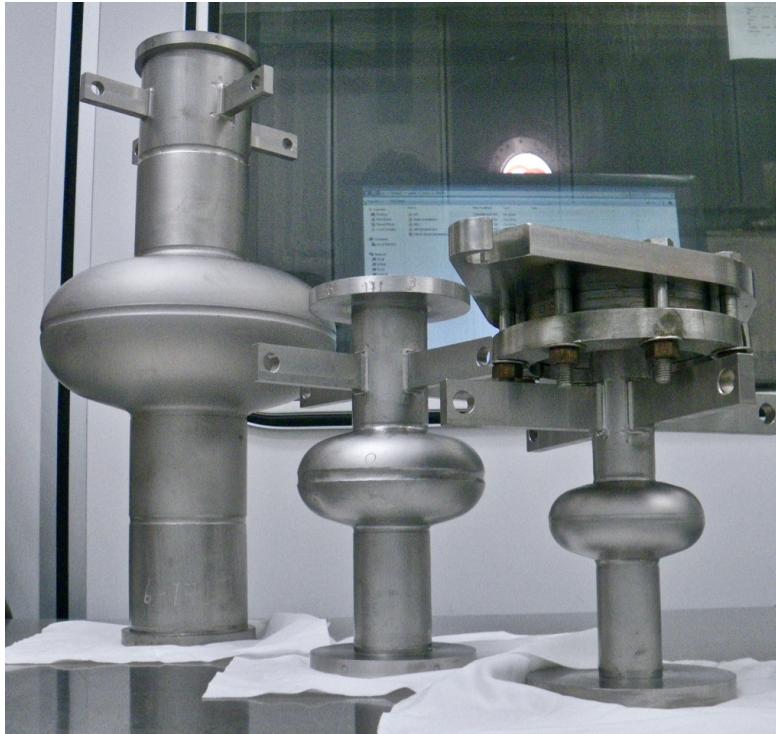
Nb<sub>3</sub>Sn forms a  
**polycrystalline** layer on  
the surface of the  
niobium



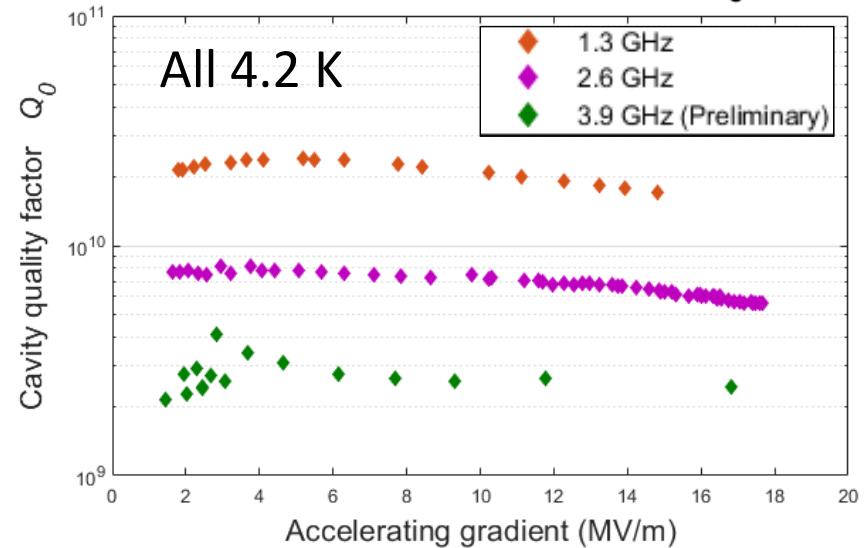
# Comparison to Niobium



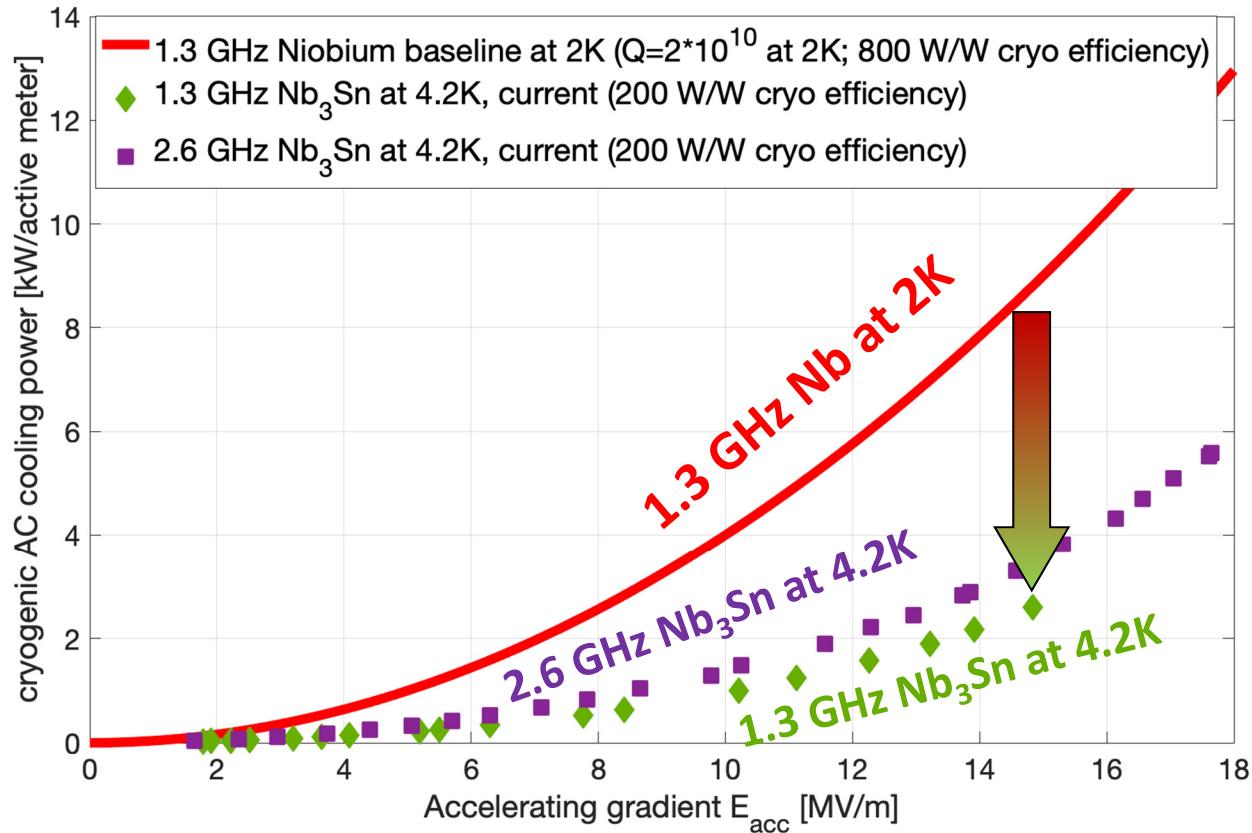
# High Frequency Nb<sub>3</sub>Sn



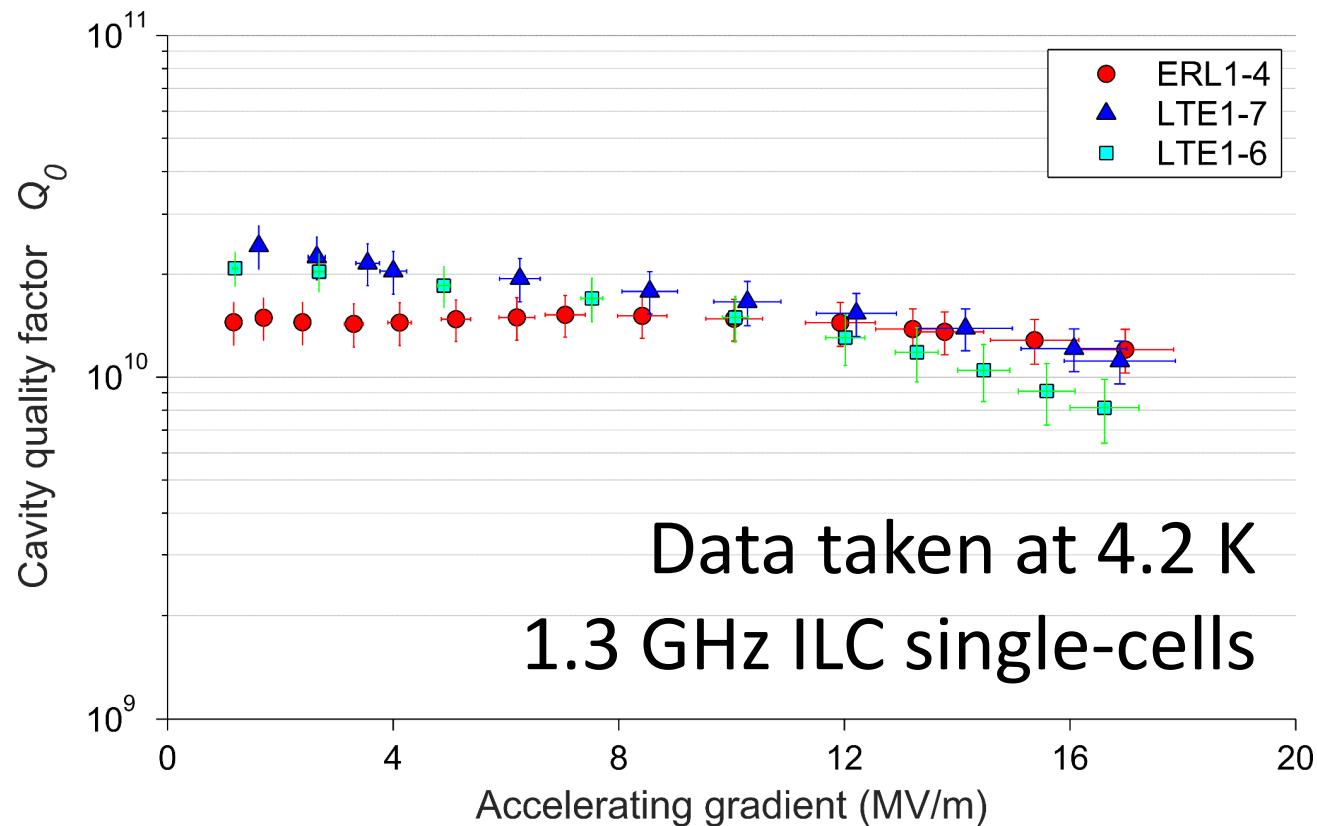
**Q vs E for Different Frequencies of Nb<sub>3</sub>Sn Cavity**



# Cryo-Efficiency



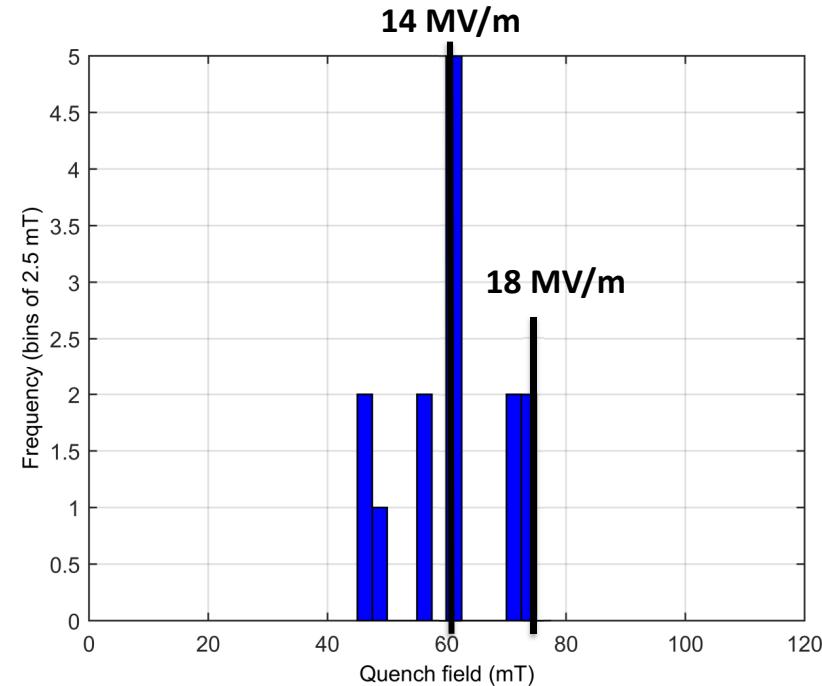
# Current performance



# Limitations in quench field

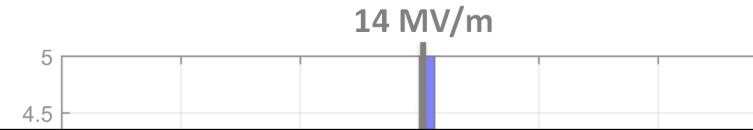
Nb<sub>3</sub>Sn cavities consistently quench at fields between  
**14 and 18 MV/m in CW operation**

The superheating  
field suggests we  
can achieve fields  
up to **96 MV/m**!



# Limitations in quench field

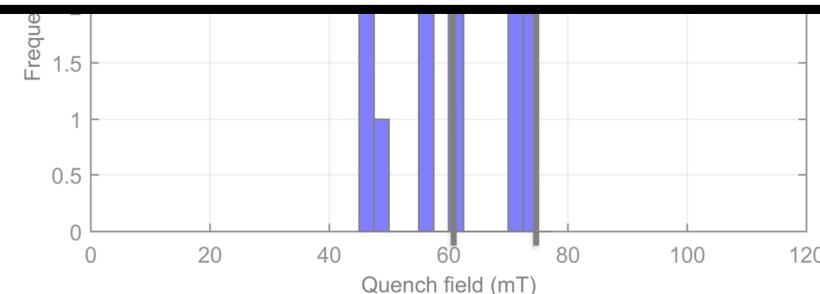
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**Conclusion:**

**Small Range of Quench Fields**



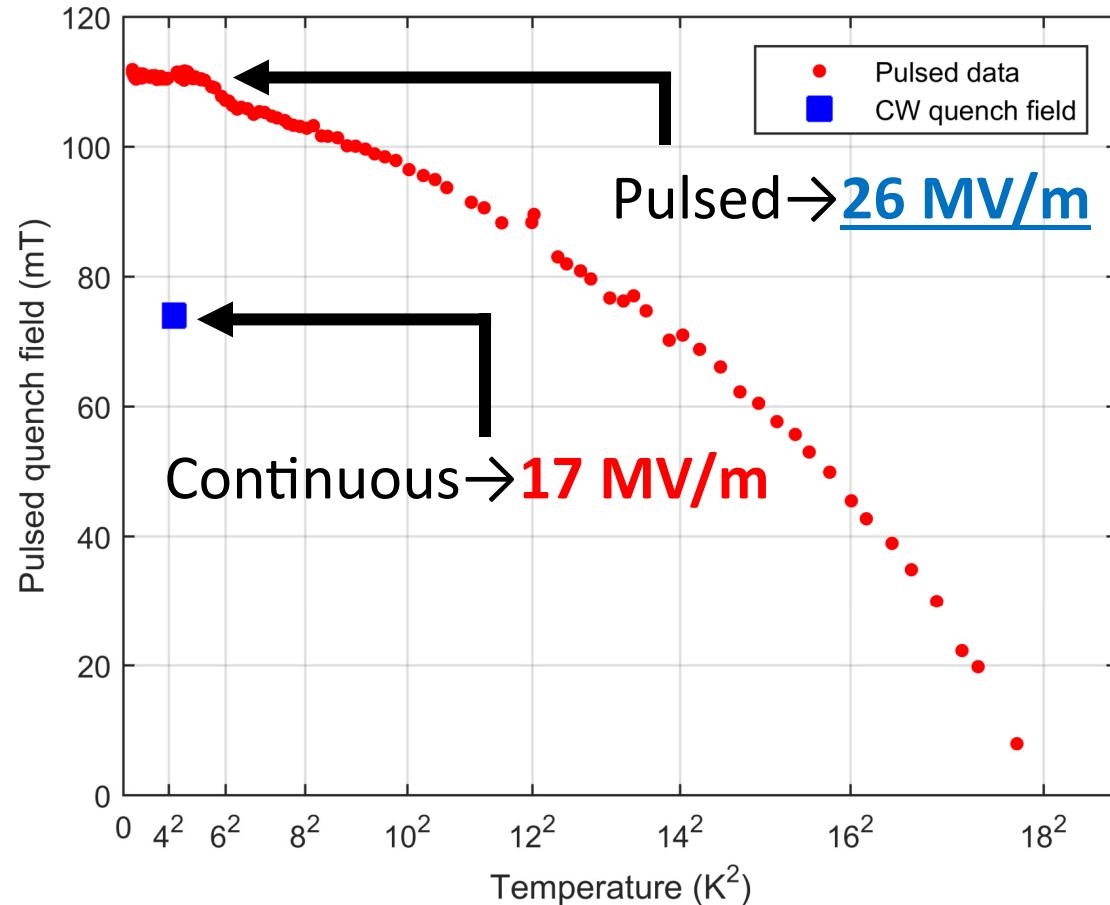


# What is limiting the quench field?



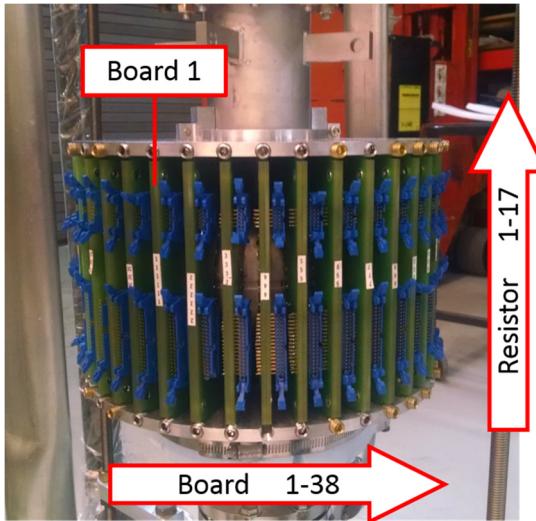
# Experimental data

# Pulsed quench field

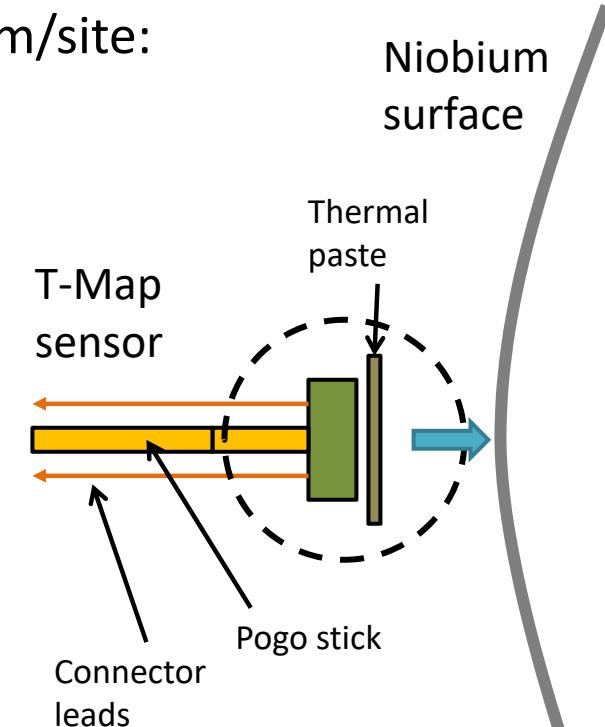
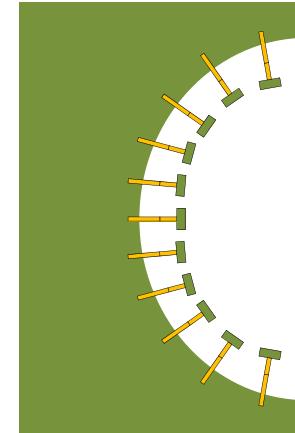


# T-Map experiment

Use temperature map to look for quench mechanism/site:

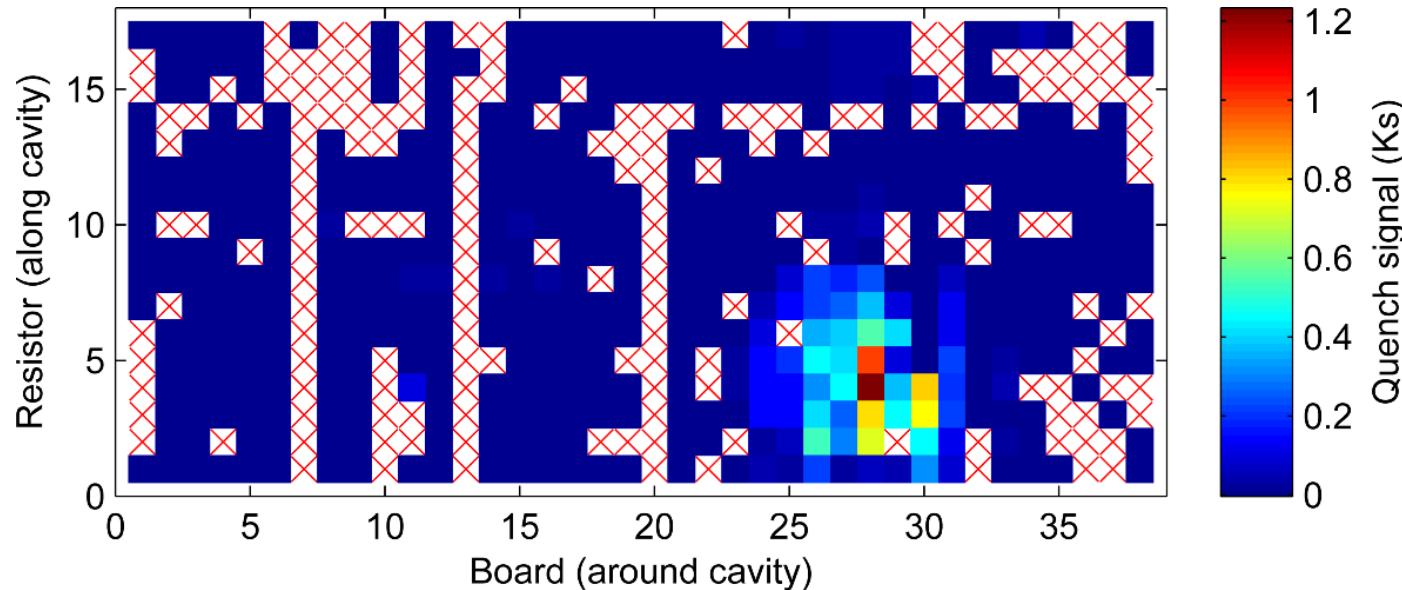


T-Map board



# Localised quench

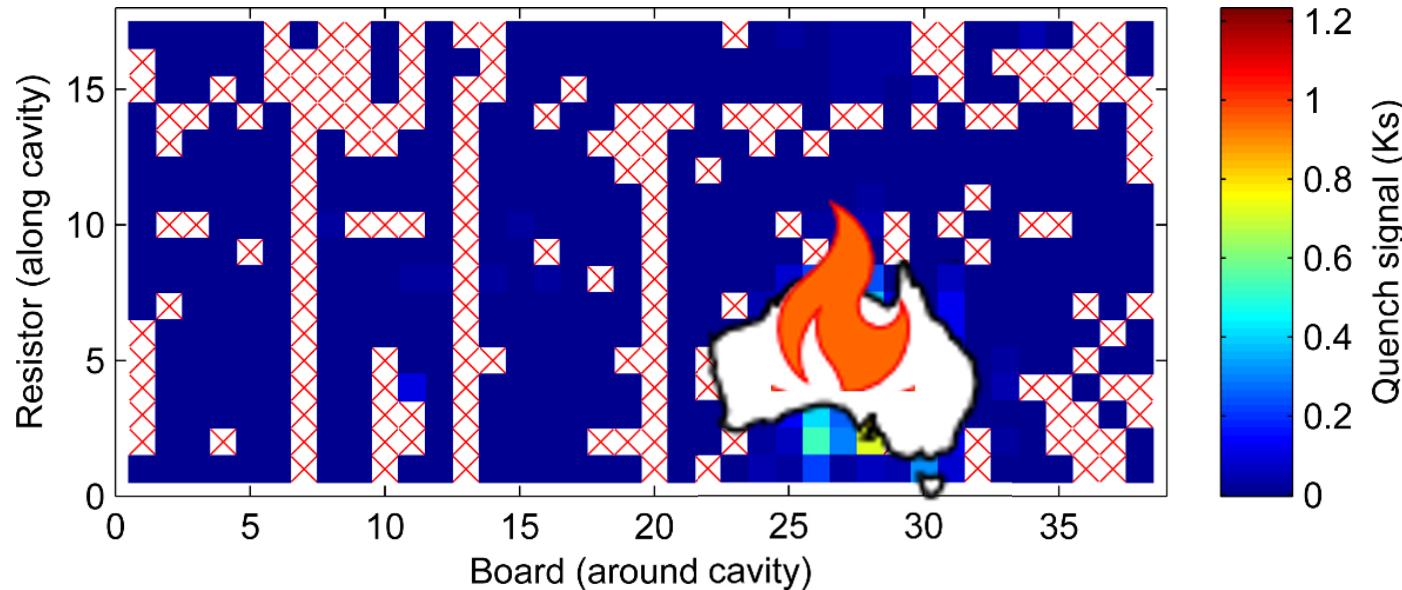
$\text{Nb}_3\text{Sn}$  cavities are limited by a quench at a localized spot



What could be at fault?

# Localised quench

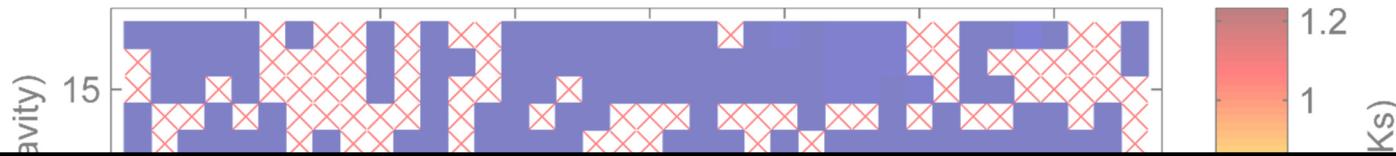
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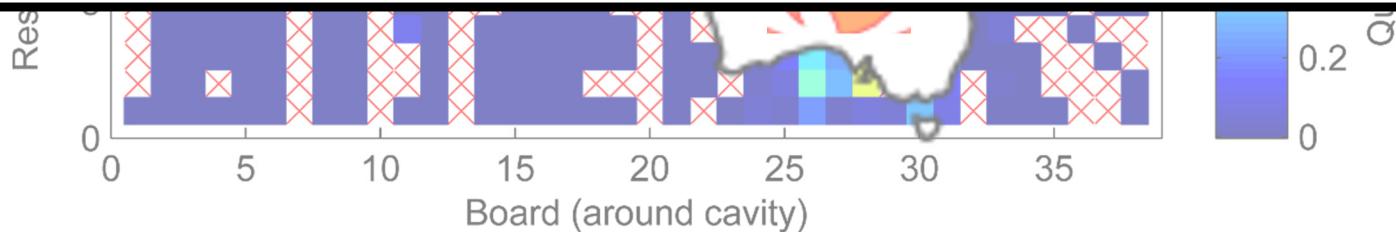
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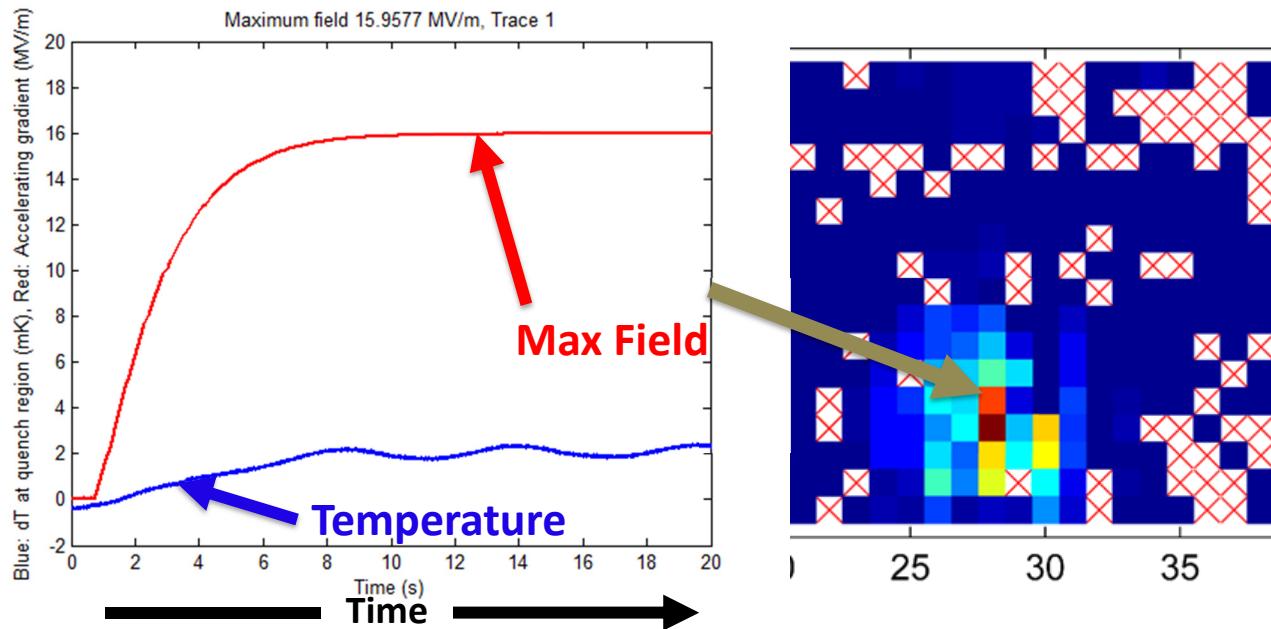
Conclusion:  
Quench is localized



What could be at fault?

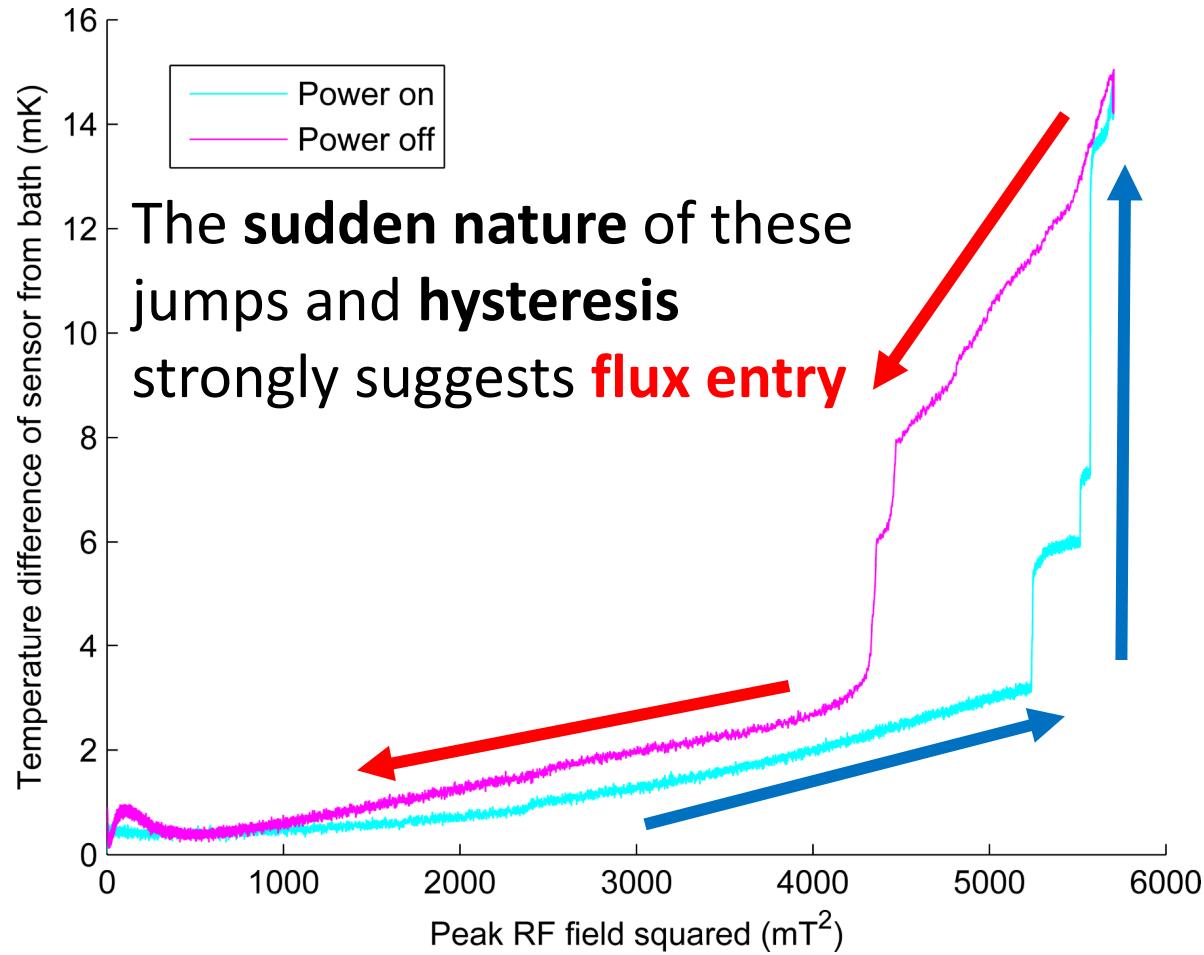
# Near quench behavior

- Measure temperature of sensor near the quench point as field is increased

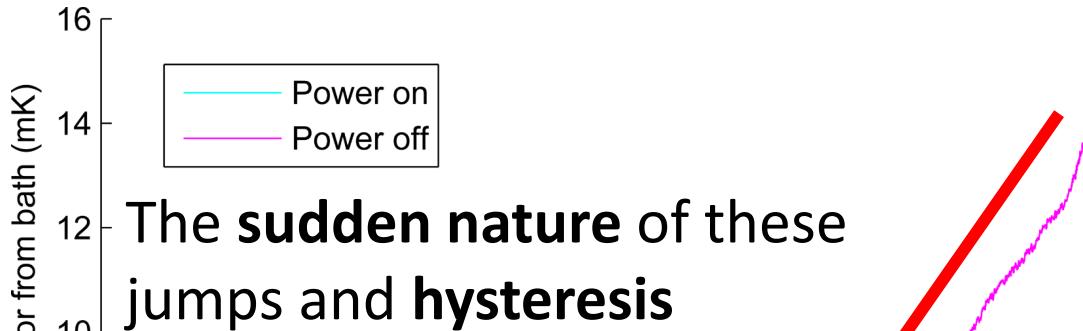


- Sudden jumps in temperature

# Near quench behaviour

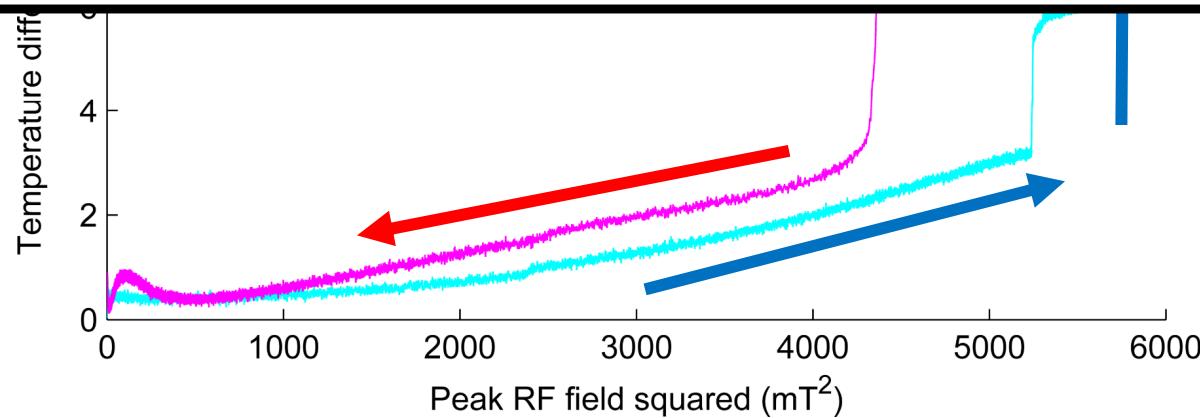


# Near quench behaviour



Conclusion:

Quench caused by vortex entry, likely at grain boundary

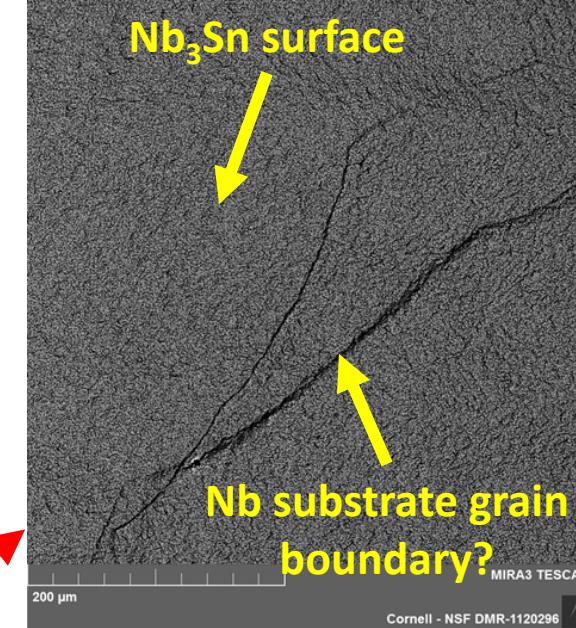


# What is happening at the surface?

- Cut out this region and examined with microscopy
- Nothing obvious except Nb grain boundary cliff
  - Rough Surface

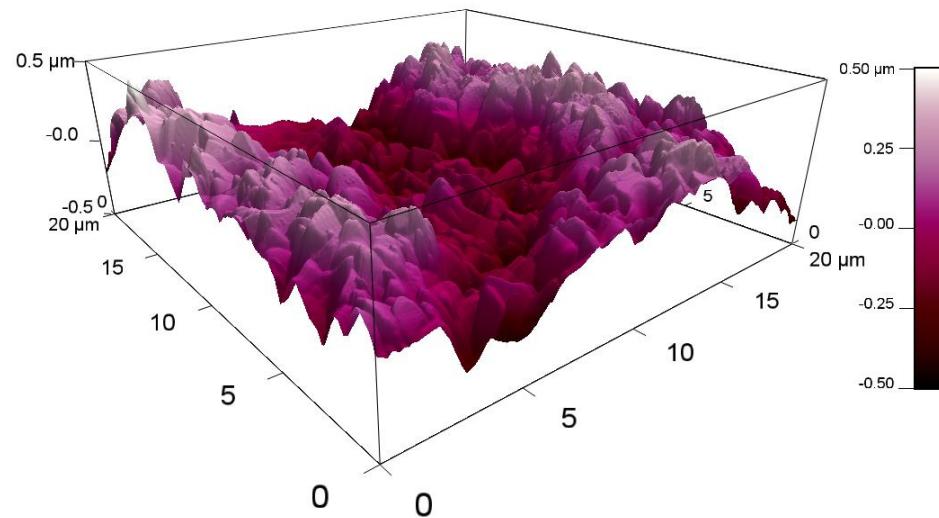


Quench Site



# Surface Roughness

- Nb<sub>3</sub>Sn we create is rougher than EP Nb
  - ~1 μm peak to peak
- This causes large enhancement of the surface magnetic field

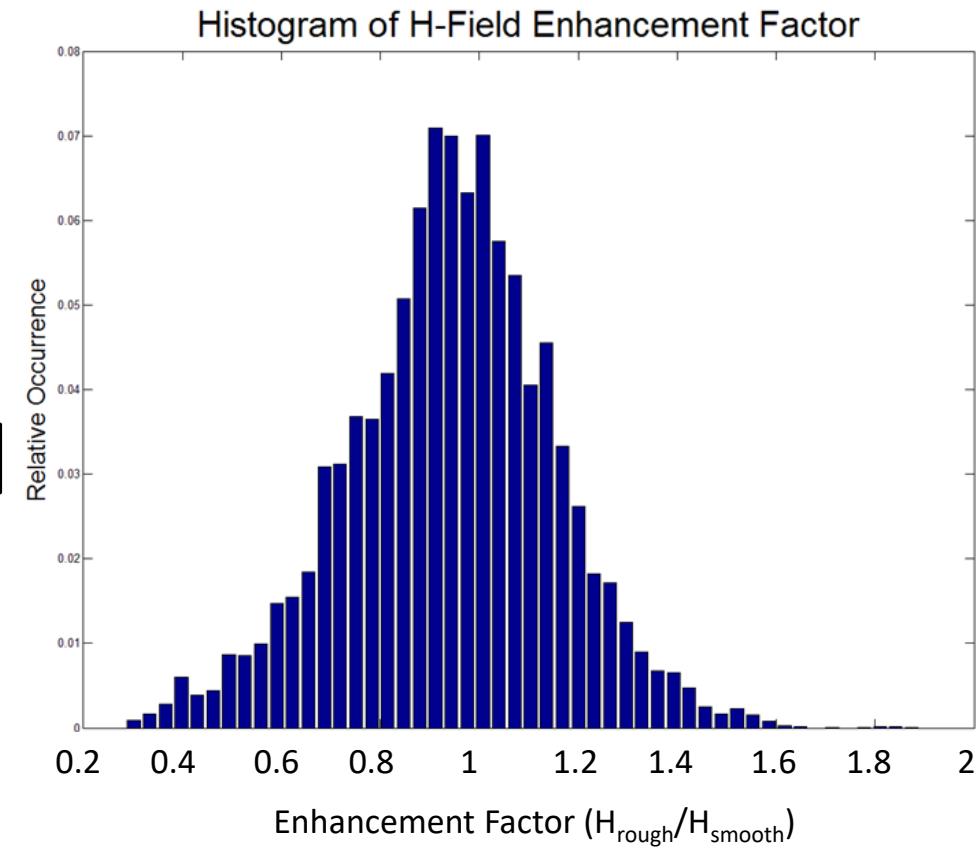


# Field Enhancement

- 1 % surface > 50% magnetic field enhancement
- Lowers defect “activation” field

17 MV/m x 1.50 field enhancement -> **25 MV/m**

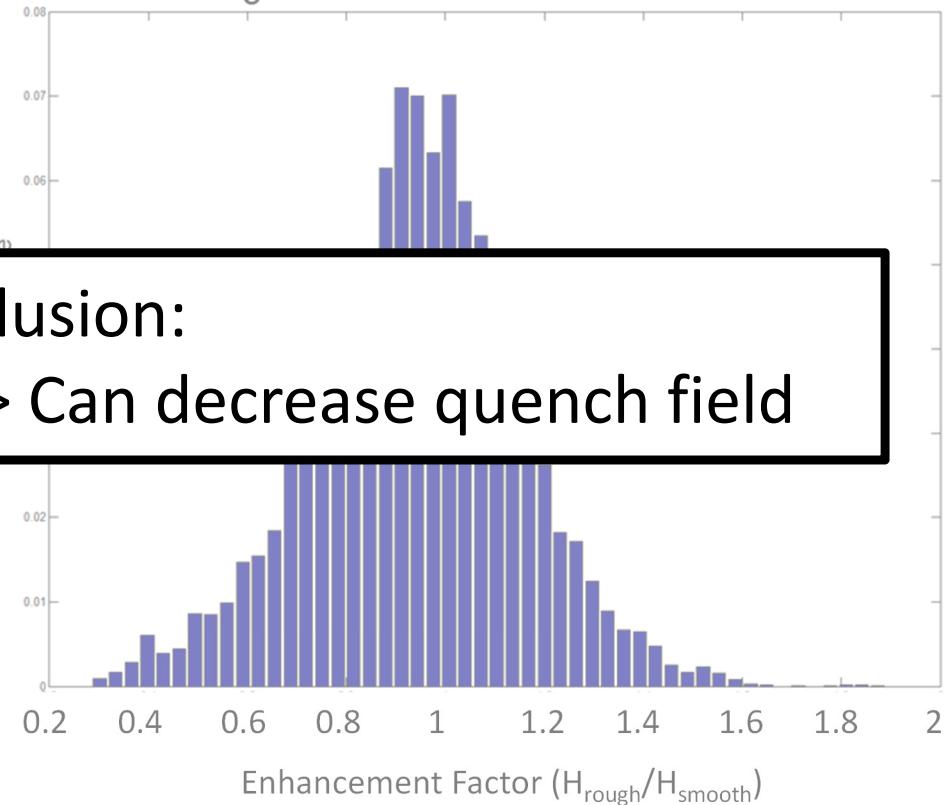
- Reducing surface roughness (growth, post-treatment) could increase quench field



# Field Enhancement

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Histogram of H-Field Enhancement Factor



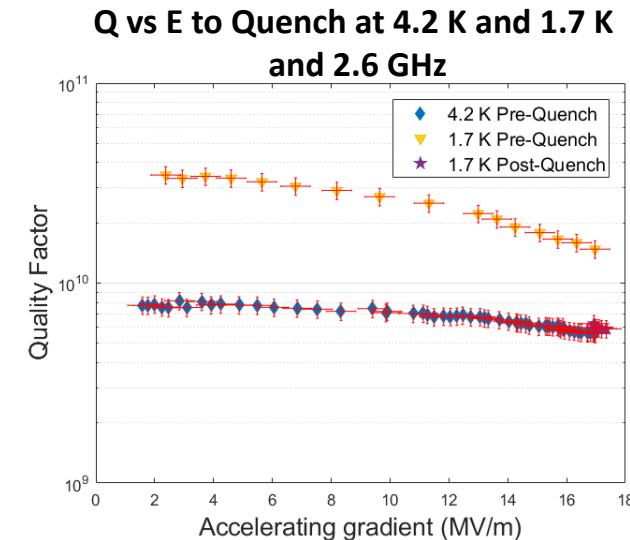
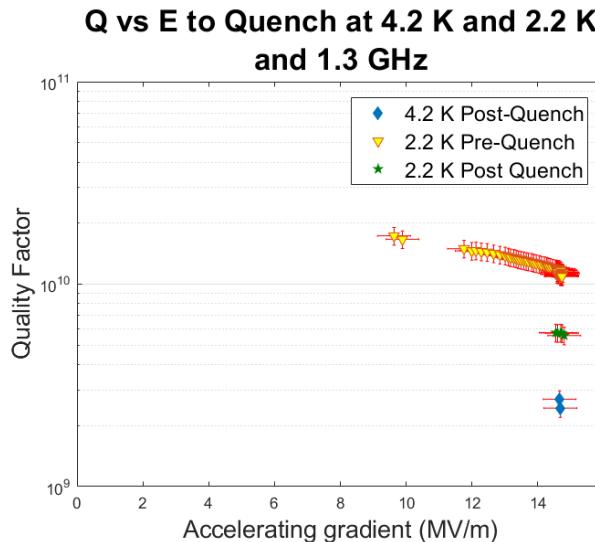
Conclusion:

High surface roughness -> Can decrease quench field

- Reducing surface roughness (growth, post-treatment) could increase quench field

# Quench Field Temperature Dependence

- Quench 4.2 K vs 2 K
  - Quench fields within ~1% (~3.4% error bar)
- No temperature dependence of quench field!
  - Limits possible quench mechanisms

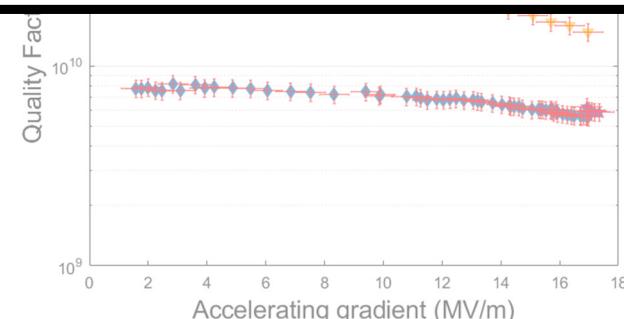
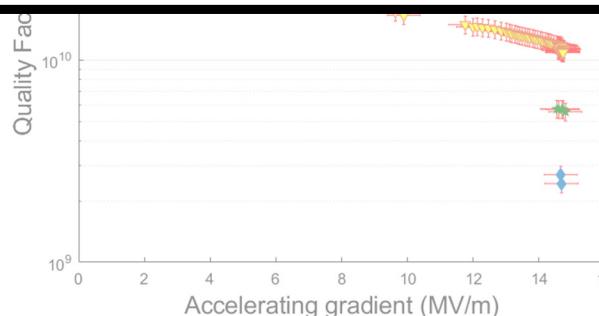


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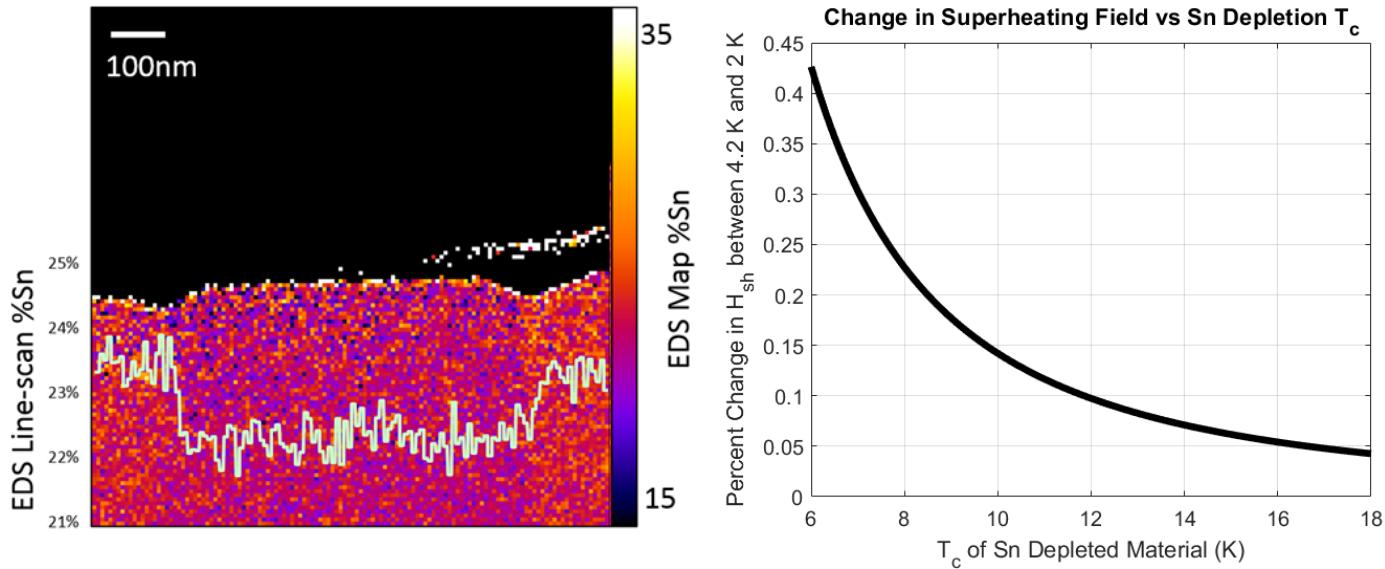
## Conclusion:

# Quench Mechanism Temperature Independent



# Quench Field Impact of Sn Depletion

- Could Sn depleted regions cause quench?
  - Quench field change with temperature **too small**

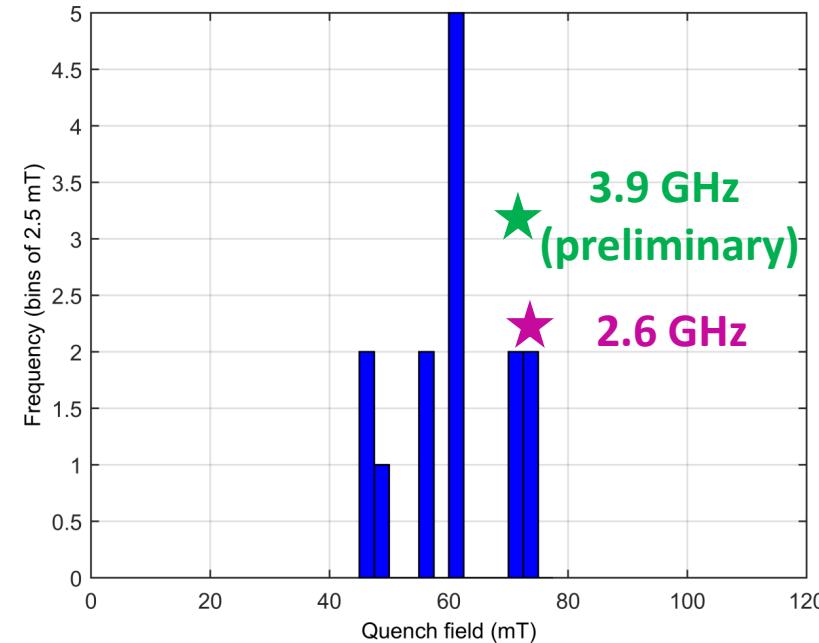


# Quench Field Frequency Dependence

The 2.6 GHz Nb<sub>3</sub>Sn cavities' quench field is consistent with 1.3 GHz Nb<sub>3</sub>Sn cavities.

Results show quench defect does not depend on frequency!

-> Limits possible defects that cause quench



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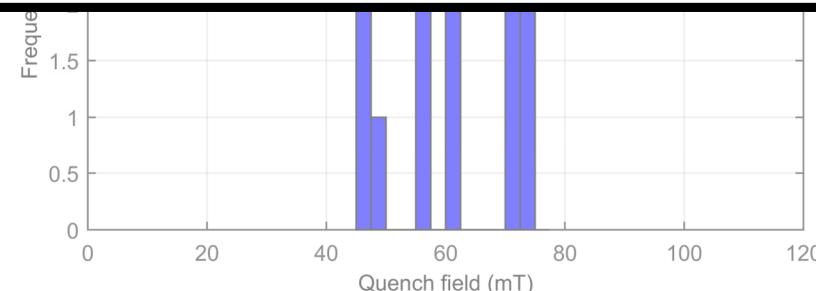
Results show



Conclusion:

Quench Mechanism Frequency Independent

-> Limits possible  
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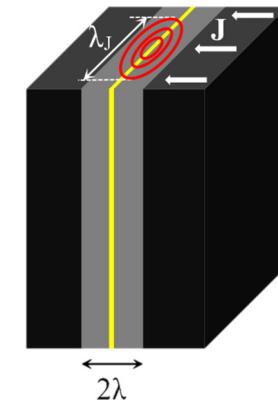
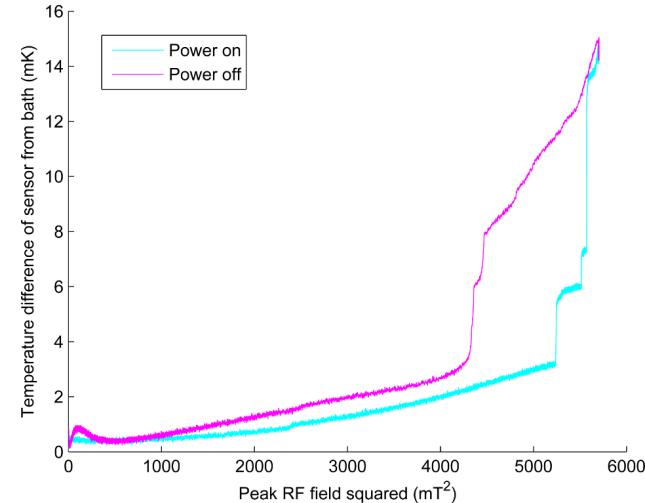
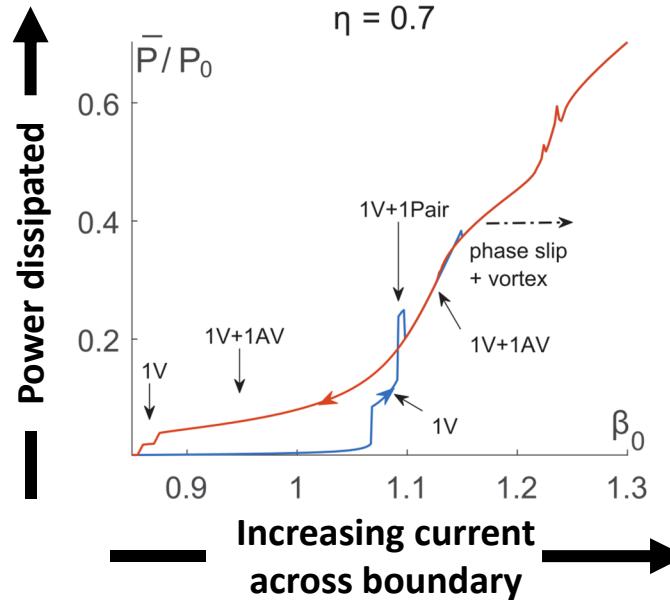
# Experimental Data Summary

- Narrow range of quench fields
- Quench localized
- Quench site warms just before quench + quantized
  - Vortex entry
- High surface roughness -> Can decrease quench field
- No temperature dependence
- No frequency dependence



# Models of Quench

# Grain boundary flux penetration



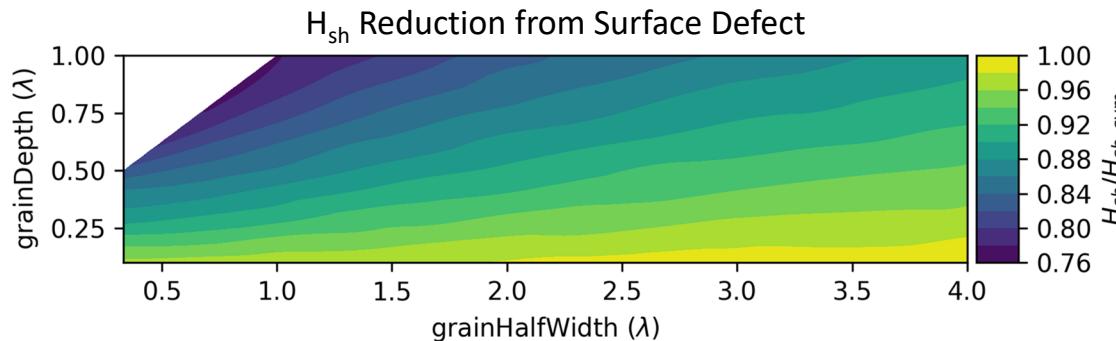
**Ahmad Sheikhzada and Alex Gurevich**  
Physical Review B **95**, 214507 (2017)  
*arXiv:1702.02843*

# Grain Boundary Flux Penetration (BYU)

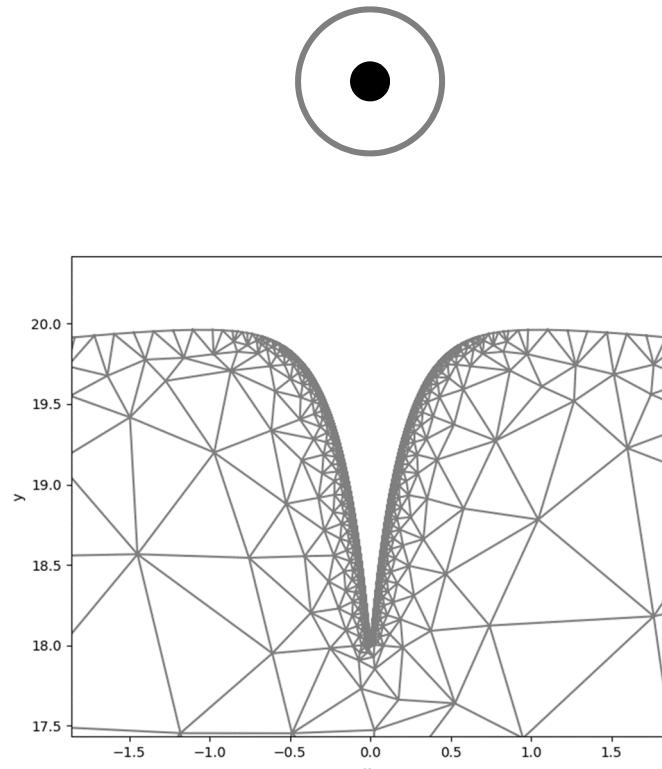


## Ginzburg-Landau Simulation of Vortex Nucleation In Grain Boundaries

- Center for Bright Beams (CBB):  
**A. R. Pack, M. Transtrum (BYU)**: MOP017
- Poor grain boundary geometry -> lower flux entry field
- Geometry effect: T, f independent\*



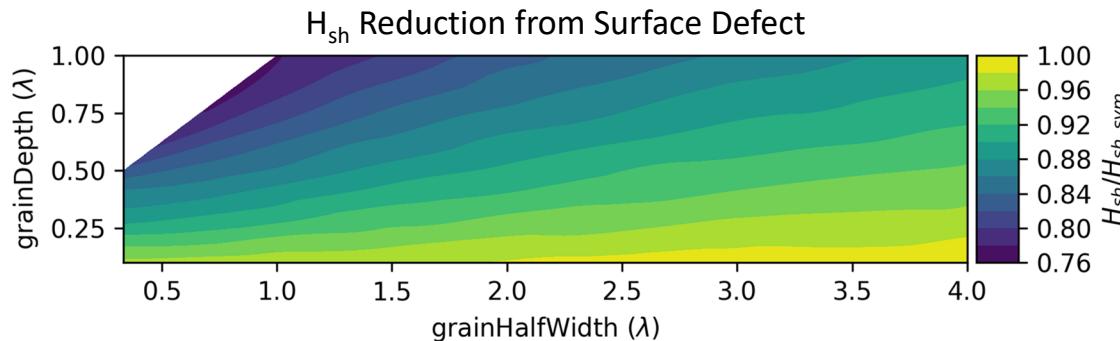
\*in cavity operating range ( $f \ll 1/T_{\text{vortex nucleation}}, T \ll T_c$ )



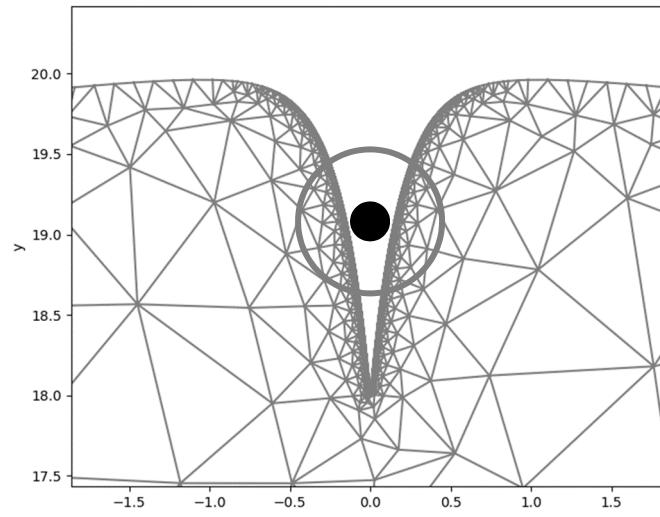
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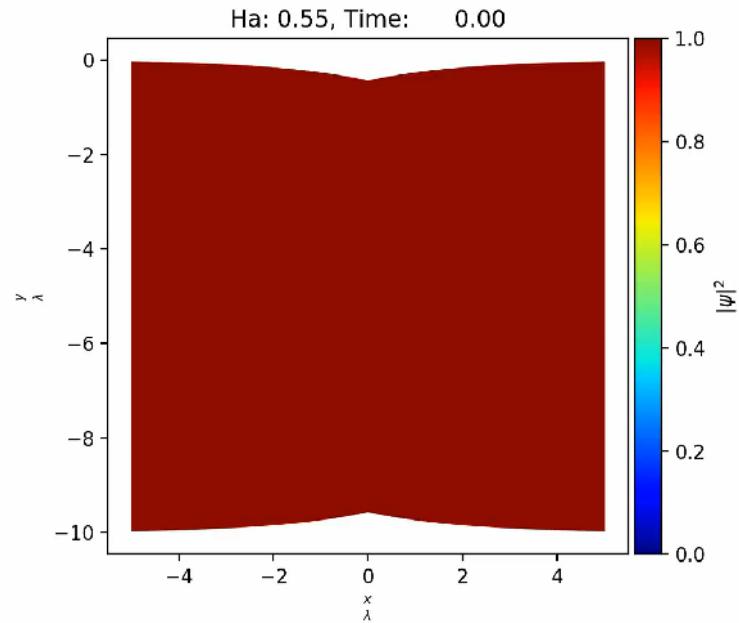


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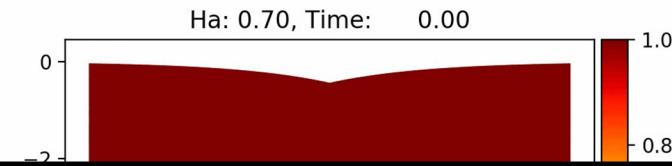
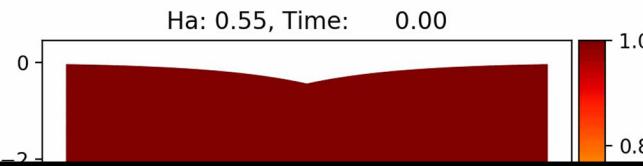
# Grain Boundary Pinning (BYU)

A. R. Pack, M. Transtrum (BYU):



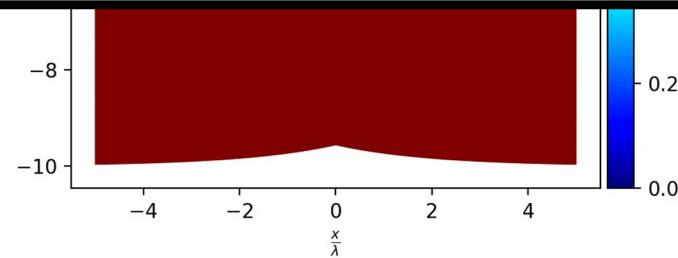
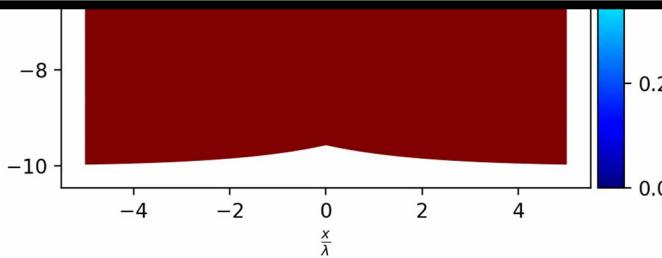
# Grain Boundary Pinning (BYU)

A. R. Pack, M. Transtrum (BYU):



Conclusion:

Grain boundary geometry/roughness lowers quench field

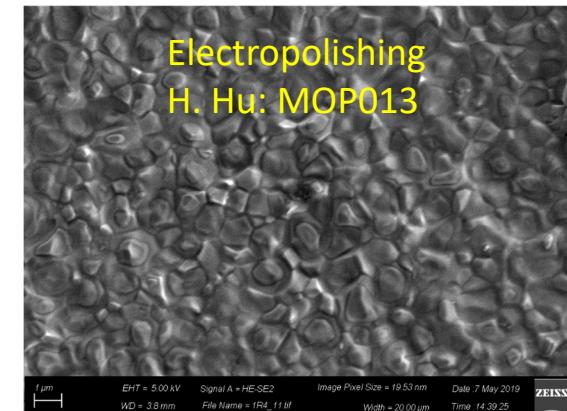
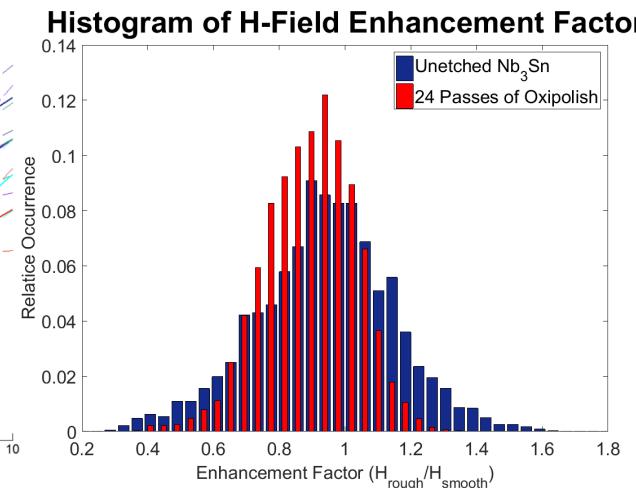
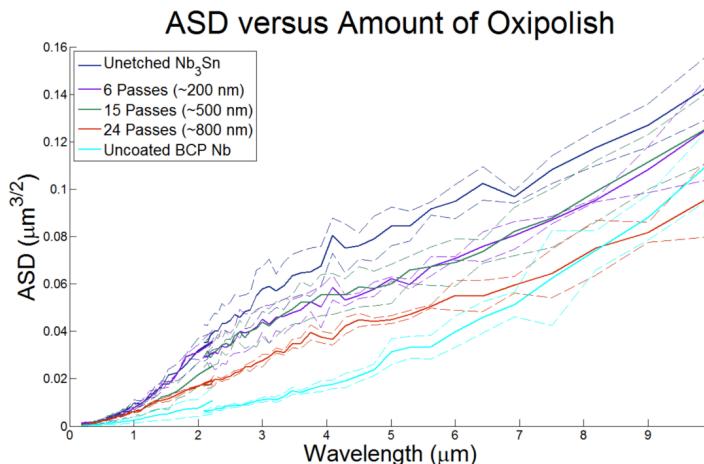




# Reducing Surface Roughness

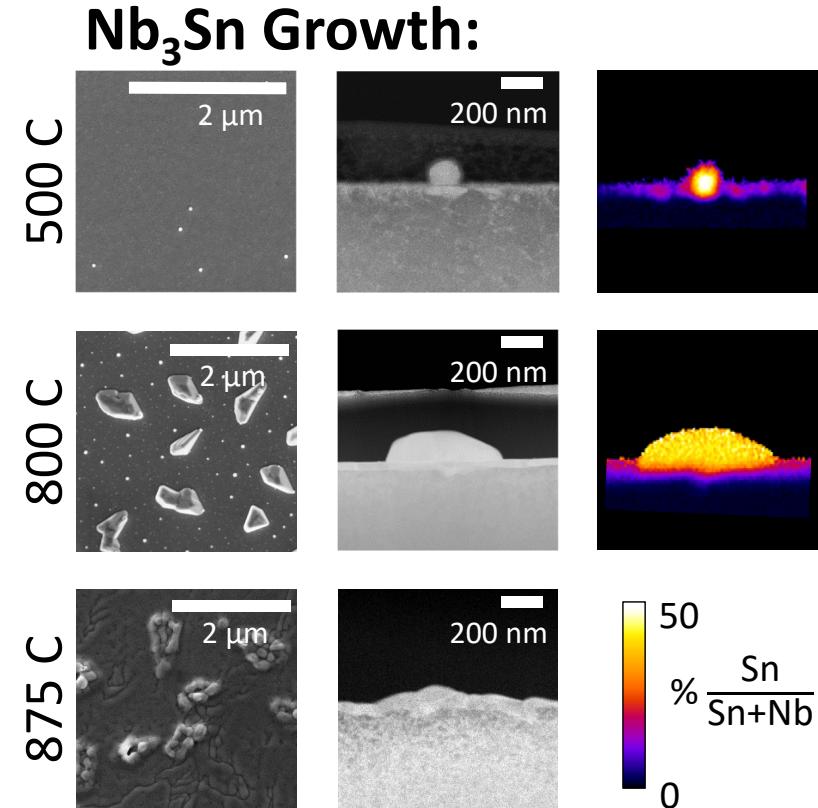
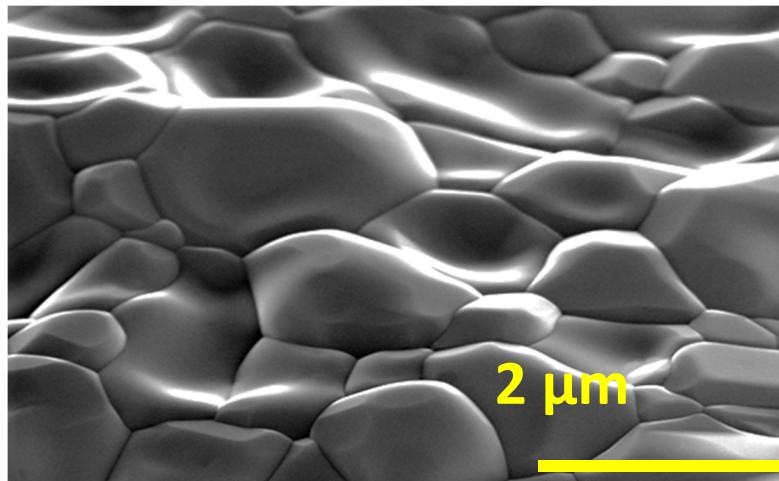
# Surface Treatment

- Developing surface treatments to reduce surface roughness
- Early result: Oxypolishing **halves roughness and surface field enhancement** with 800 nm removal



# Why is Nb<sub>3</sub>Sn Rough?

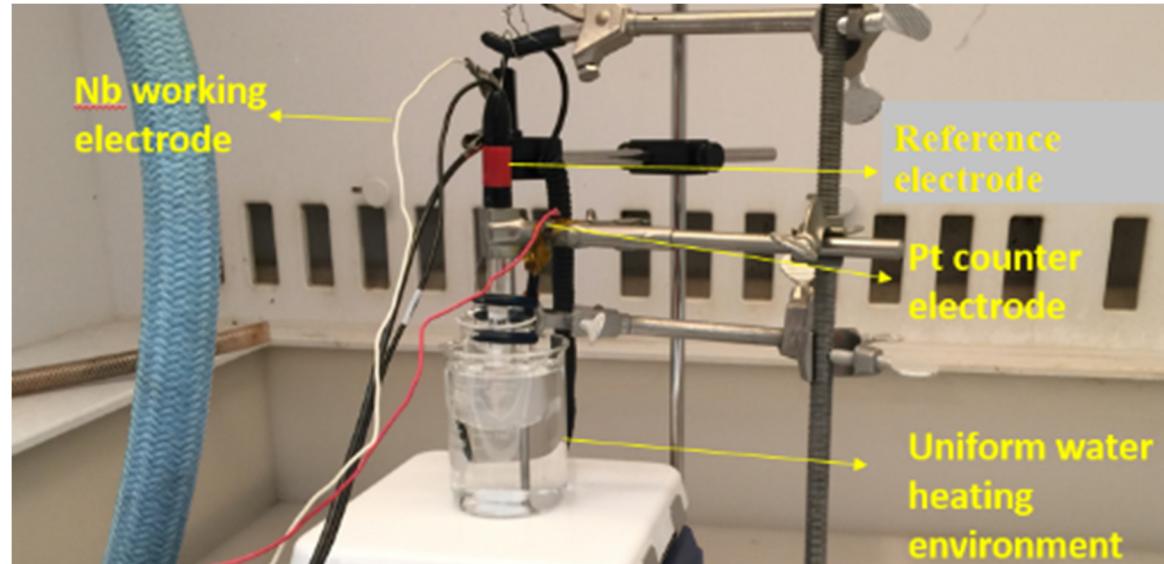
- Nb<sub>3</sub>Sn roughness comes from growth
  - Bad Sn nucleation -> **rough surface**
  - Good Sn nucleation -> **smooth surface**



# Sn Electroplating

Zeming Sun (Cornell):

- Electroplate Sn onto Nb before heat treatment
  - > Grow smoother  $\text{Nb}_3\text{Sn}$





# Sn Electroplating

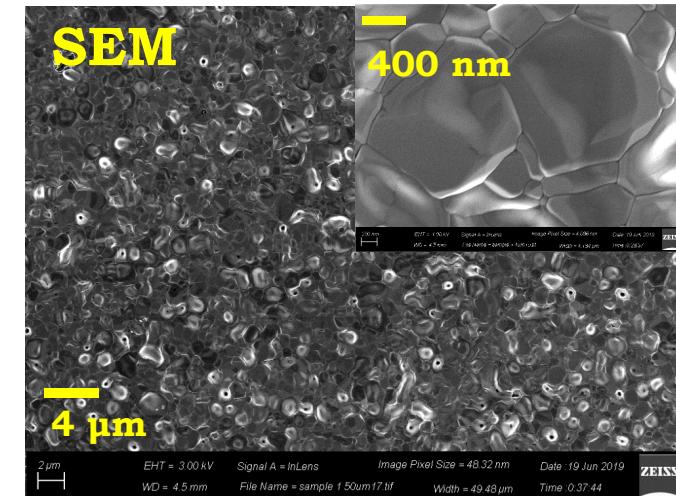
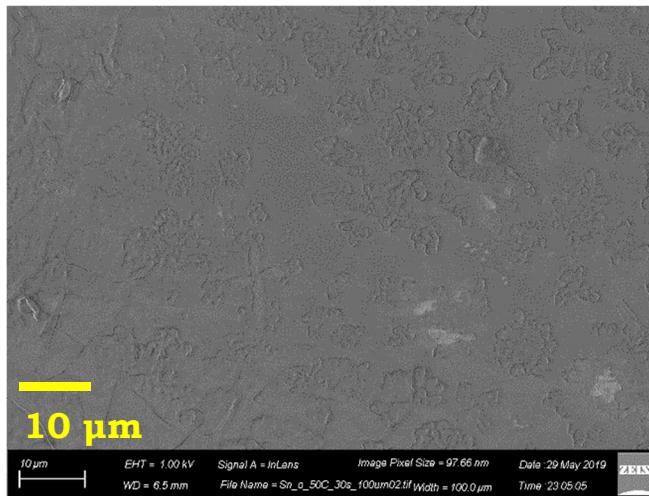
Coated Sn



$Nb_3Sn$



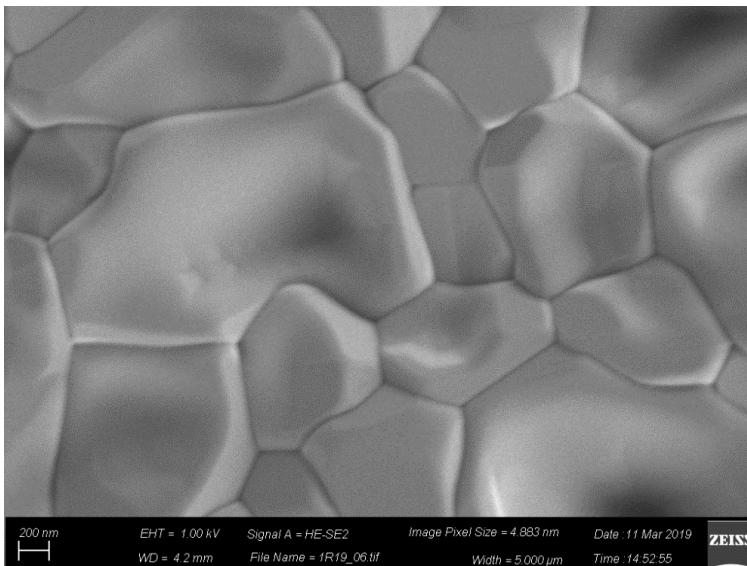
Heat Treatment



# Sn Electroplating

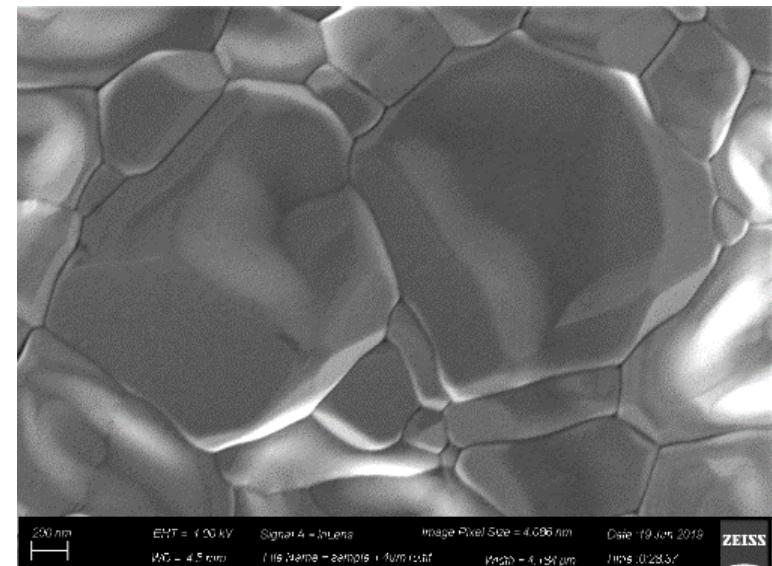
Sn<sub>2</sub>Cl Nucleation

R<sub>a</sub> ~ 300 nm



“Sn Plating Nucleation”

R<sub>a</sub> ~ 60 nm

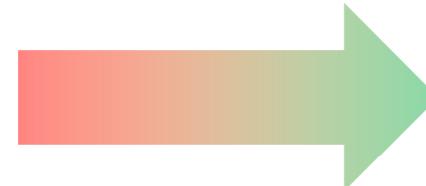


Next step: Grow entire cavity using Sn plating

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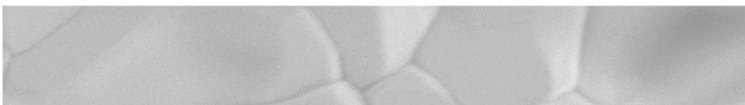
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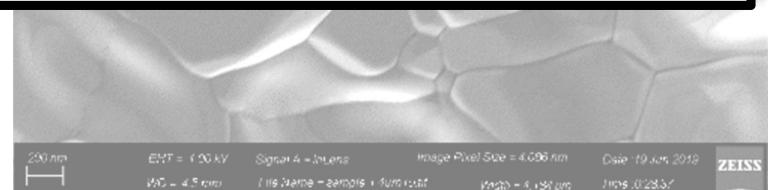
“Sn Plating Nucleation”

R<sub>a</sub> ~ 60 nm



Conclusion:

Sn plating nucleation 5 x roughness reduction!



Next step: Grow entire cavity using Sn plating



# Conclusions

- From experiment:
  - Claim: Vortex entry at grain boundaries a likely quench mechanism
- Reducing surface roughness is a critical next step
- Can grow smoother Nb<sub>3</sub>Sn with Sn plating



# Acknowledgements

The Cornell Nb<sub>3</sub>Sn program is supported by:

U.S. DOE award DE-SC0008431: 1.3 GHz Nb3Sn tests and Nb3Sn R&D

NSF Award 1734189: 2.6 GHz and 3.9 GHz tests and R&D

Center for Bright Beam (NSF Award 1549132): Materials studies

This work make use of Cornell Center for Materials Research, NSF MRSEC program (DMR-1719875)

with special thanks to

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Dr. Zeming Sun

James Sears

Prof. Tomas Arias

Dr. Daniel Hall

Greg Kulina

Prof. David A. Muller

Paul Cueva

John Kaufman

Prof. James P. Sethna

Nathan Sitaraman

Holly Conklin

Prof. Mark Transtrum

James Maniscalco

Terri Gruber

Dr. Danilo Liarte

Alden Pack

Paul Bishop