

Material & Superconducting Properties of NbTiN/AlN Multilayer Films

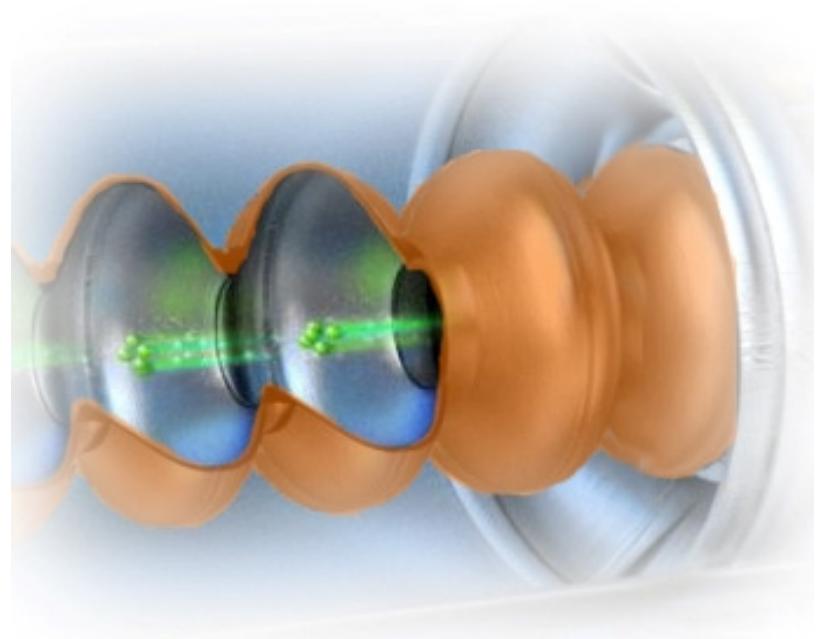
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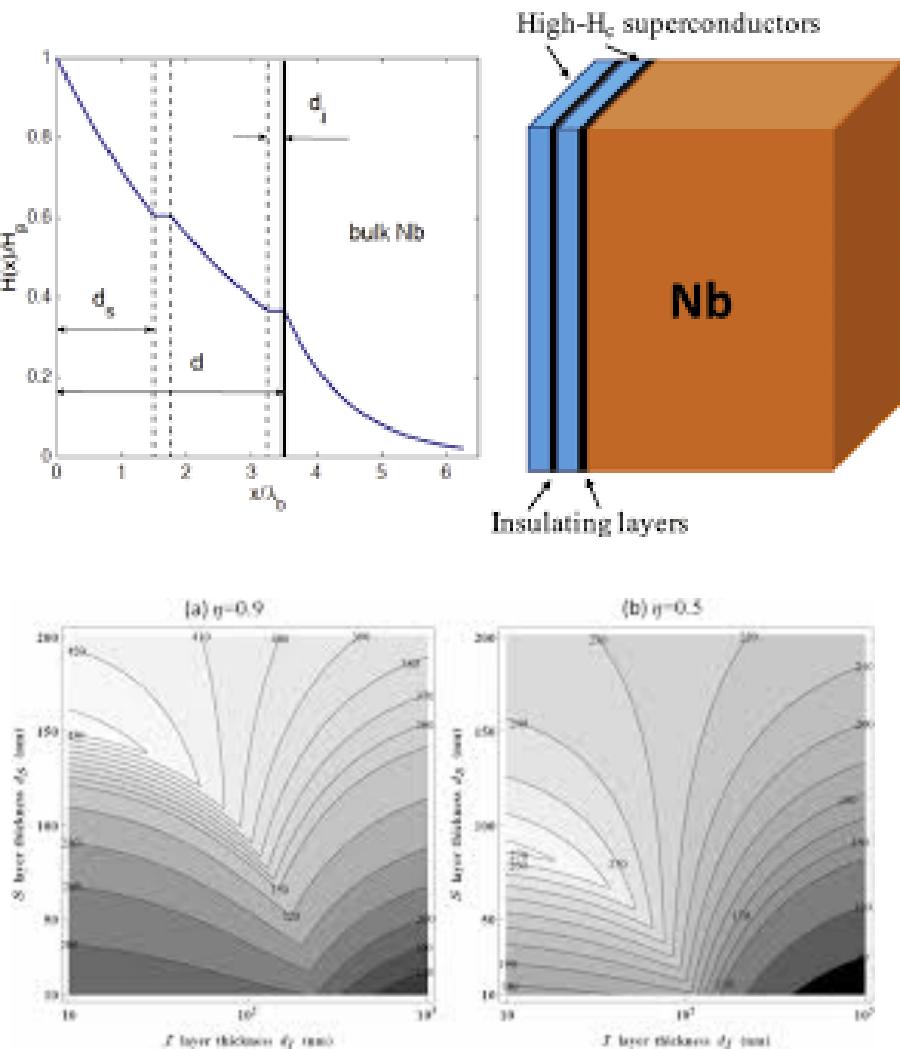


OUTLINE

- ❑ Introduction
- ❑ Materials choice – reminders
- ❑ Experimental Setup
- ❑ NbTiN based structures on niobium
- ❑ QPR Results – summary
- ❑ NbTiN based QPR structures
- ❑ Conclusion & Future Work



SRF Application beyond Nb: SIS Multilayers



Taking advantage of the high $-T_c$ superconductors with much higher H_c without being penalized by their lower H_{c1} ...

Alex Gurevich, Appl. Phys. Lett. 88, 012511 (2006)

Alex Gurevich, AIP ADVANCES 5, 017112 (2015)

T. Kubo, Applied Physics Letters 104, 032603 (2014)

**Multilayer coating of SC cavities:
alternating SC and insulating layers with $d < \lambda$**
**Higher T_c thin layers provide magnetic screening
of the Nb SC cavity (bulk or thick film) without
vortex penetration**

- Strong increase of H_{fp} in films allows using RF fields $> H_c$ of Nb, but lower than those at which flux penetration in grain boundaries may become a problem => no transition, no vortex in the layer
- High H_{fp} , applied field is damped by each layer
- Insulating layer prevents Josephson coupling between layers
- Applied field, i.e. accelerating field can be increased without high field dissipation
- SC layers with higher T_c , Δ (Nb_3Sn , NbN , etc.) => Strong reduction of R_{BCS} (ie high Q_0)

Possibility to move operation from 2K to 4.2K

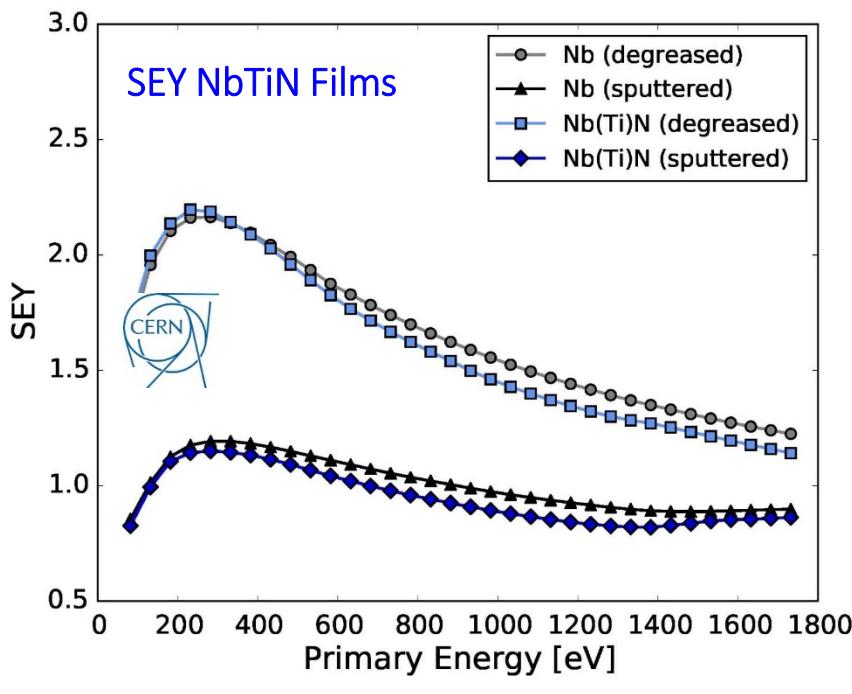
SIS Multilayers: Materials Choice - Reminder

NbTiN

Good superconductor, bulk $T_c \sim 17.3$ K

δ -phase for $T_c > 16$ K

Typically more stable and metallic behavior than NbN

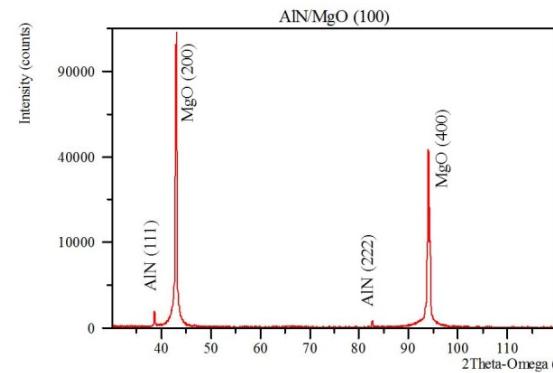


Max. SEY = 2.2 ± 0.1 comparable to EP Nb

After sputtering away ~ 3 nm, SEY down to 1.15

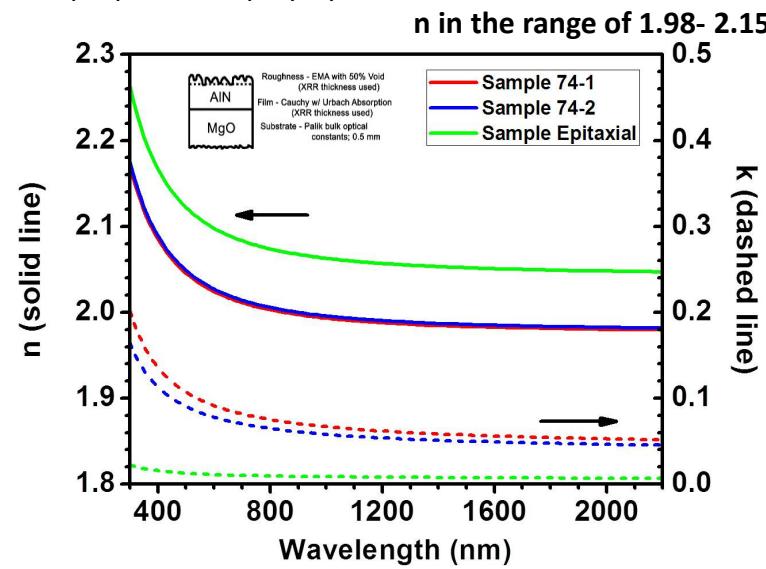
AlN

Good Dielectric Behavior



Good quality AlN are readily produced at 600 and 450°C by dc-reactive magnetron sputtering with $N_2/Ar \sim 33\%$. The films exhibit the cubic structure (single crystal) at 600 °C and the hexagonal structure (polycrystalline) at 450 °C.

At 450 °C, 30 nm AlN films exhibit dielectric properties of polycrystalline AlN films



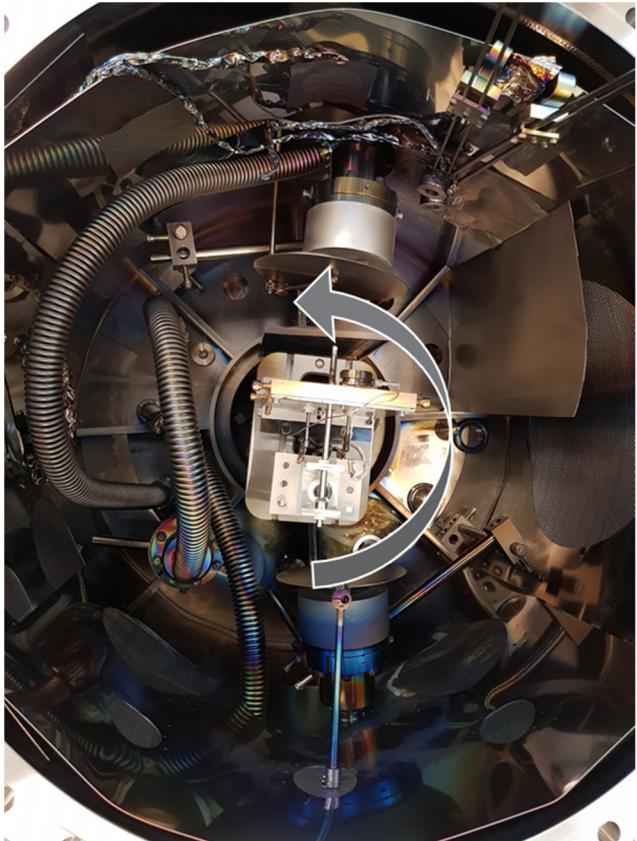
UHV DEPOSITION SYSTEM upgrade

Samples heated up to 600 °C

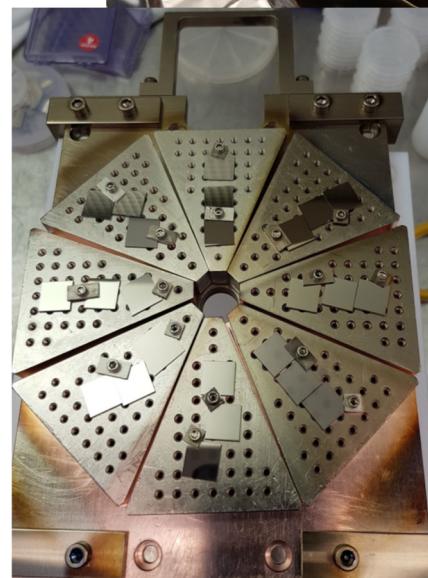
Base pressure 10^{-10} Torr

3 magnetron sources

DC, RF and HiPIMS capable

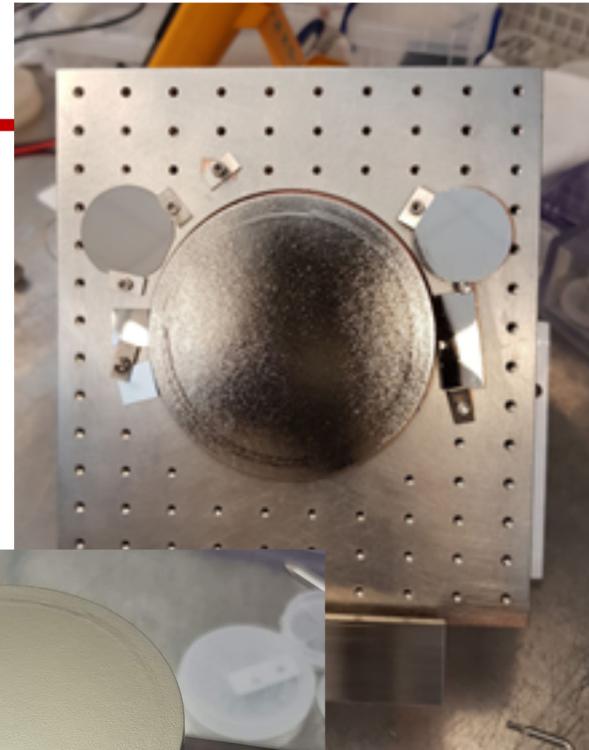
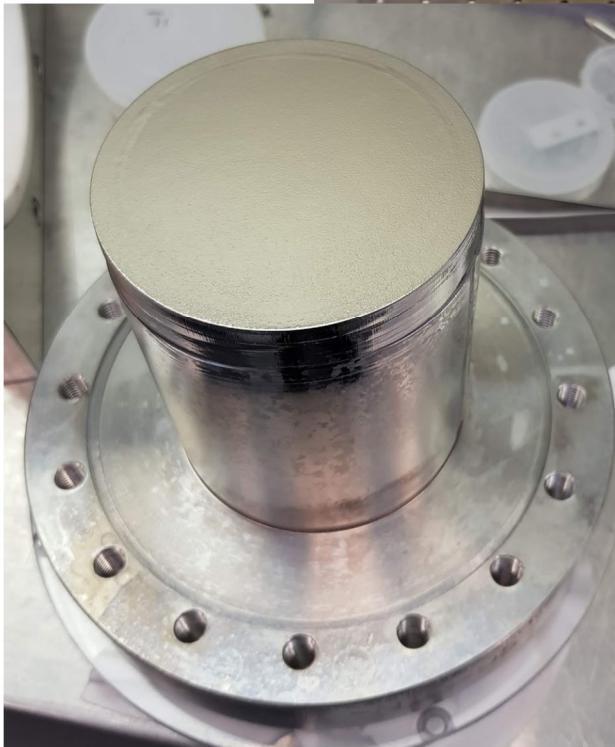
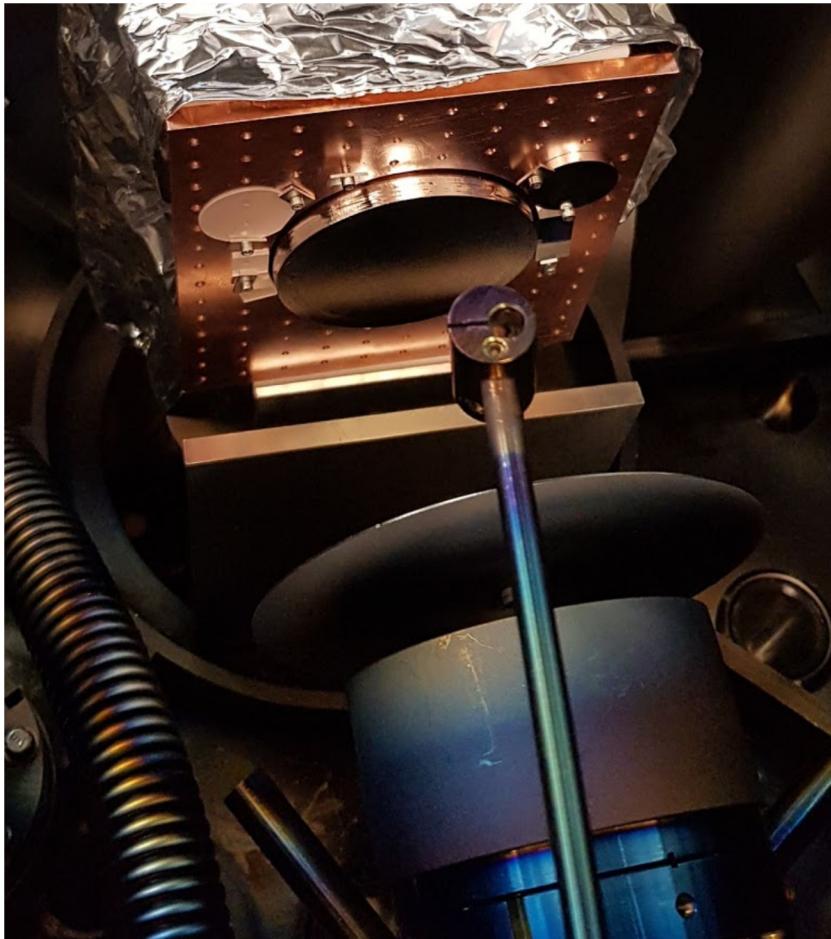


Central stage
mounted on
differentially
pumped CF system
to position samples
in front of each
magnetron source



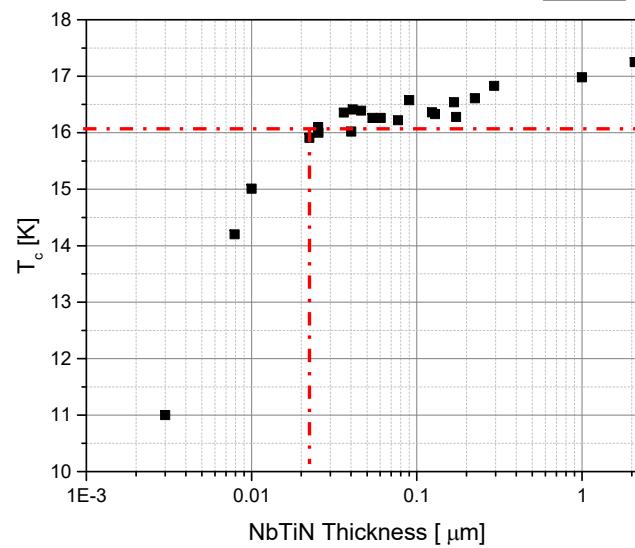
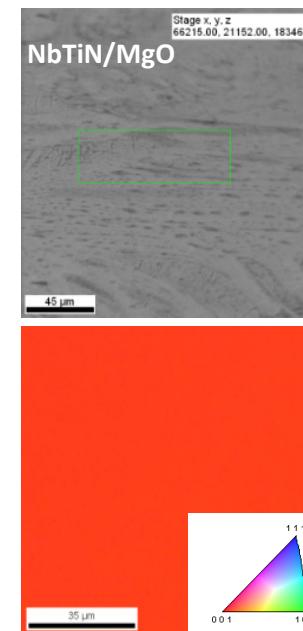
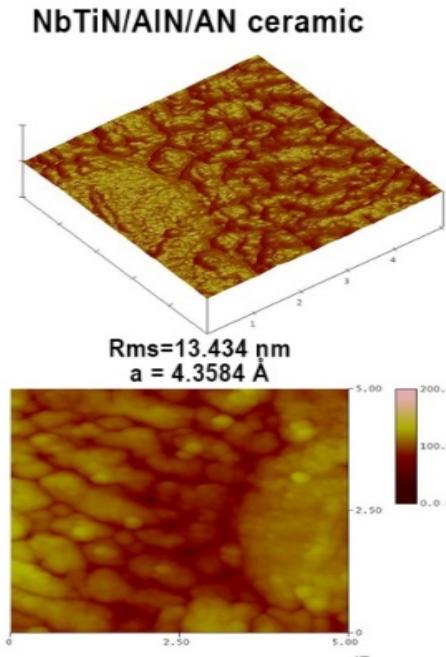
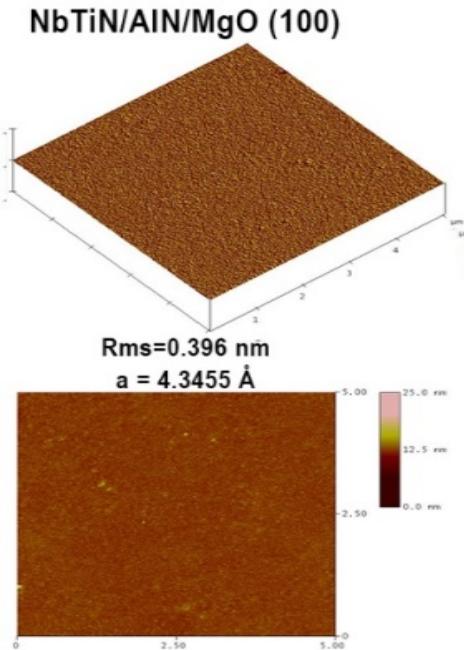
UHV DEPOSITION SYSTEM

QPR Sample Coating Setup



Samples are mechanically polished, heat treated at 800 °C and electropolished

SIS NbTiN/AlN structures on MgO & Nb surfaces

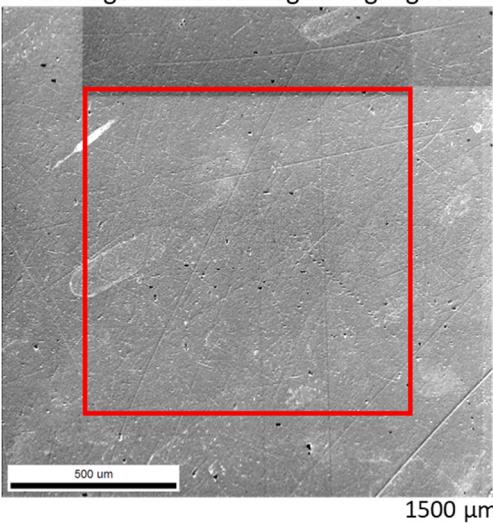


NbTiN/MgO
~ ideal case

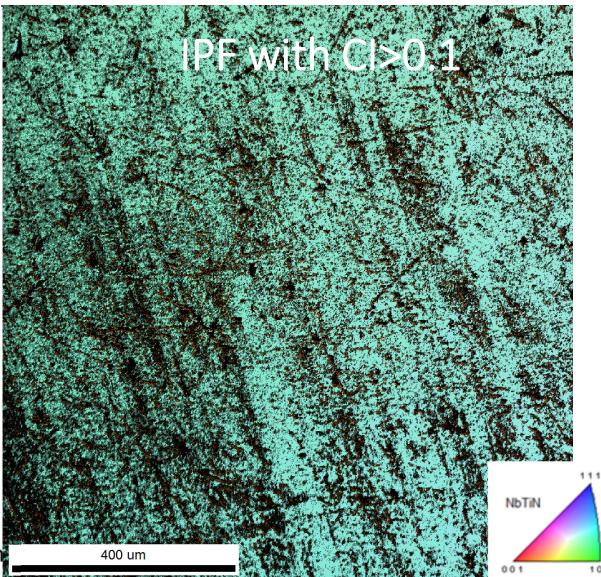
NbTiN, NbTiN/AlN ceramic
Have a T_c slightly depressed
in comparison

Microstructure of NbTiN Films on Nb

SE Image with EBSD Region Highlighted

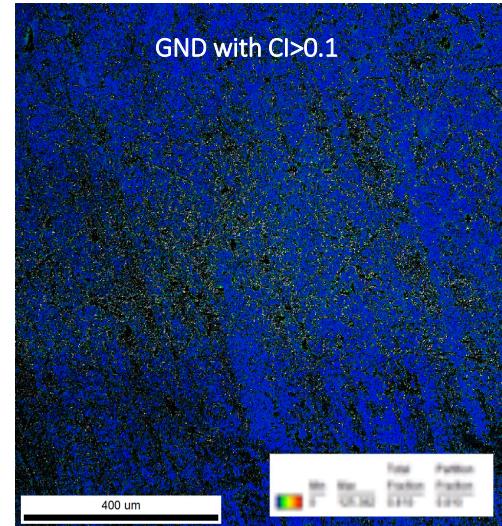


IPF with $C_l > 0.1$



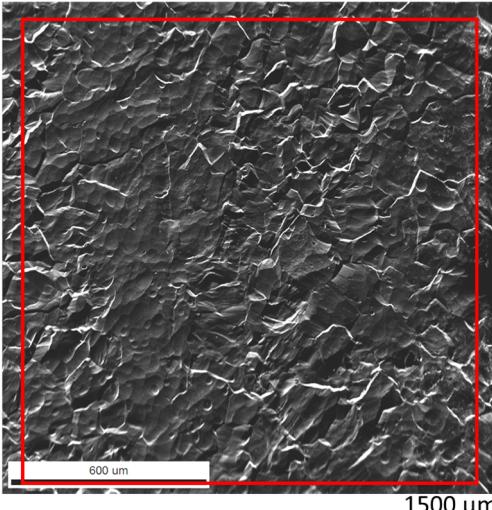
2 μm “bulk” NbTiN/ single crystal Nb

GND with $C_l > 0.1$

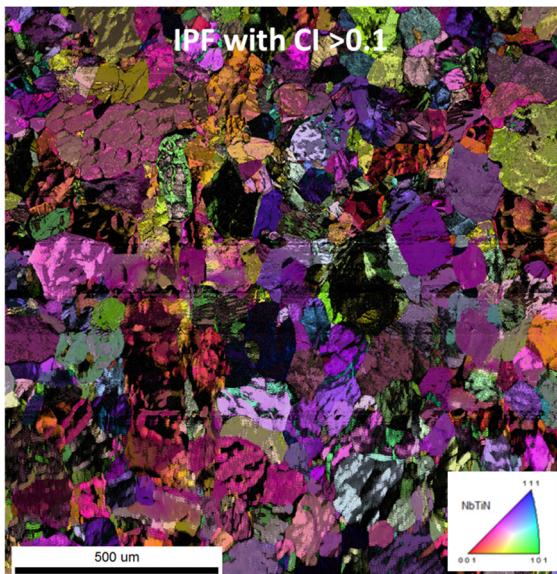


$T_c = 17 \text{ K}$

SE Image with EBSD Region Highlighted

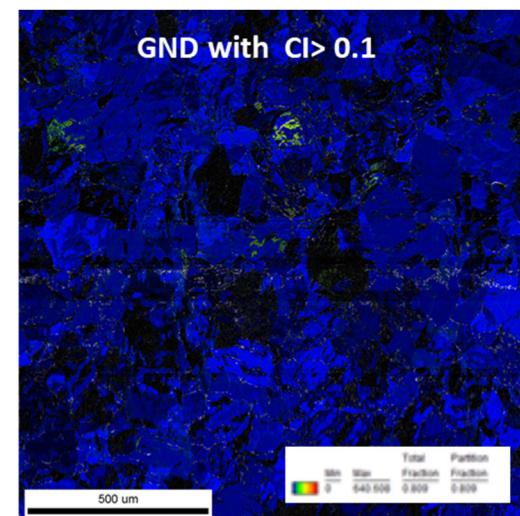


IPF with $C_l > 0.1$



200 nm NbTiN/Nb

GND with $C_l > 0.1$



$T_c = 14.8 \text{ K}$

Film Interface development in SIS Multilayers - Metamaterials

Metamaterial superconductor based on NbTiN (metamaterials synergistic project, DARPA-BAA funded)

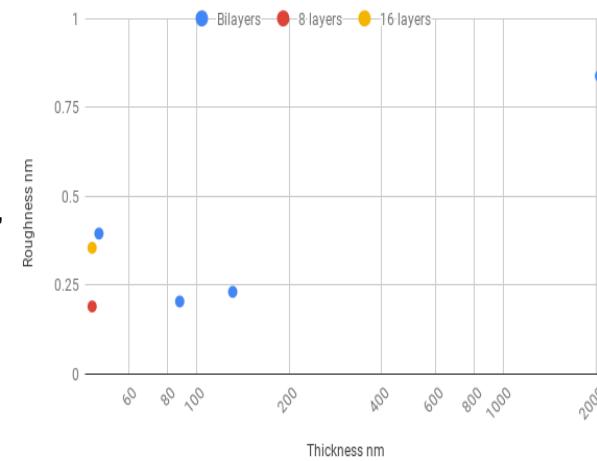
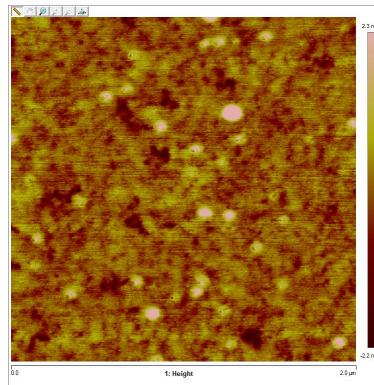
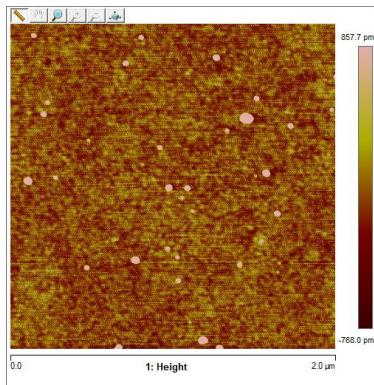
Nature Scientific Reports 6, Article number: 34140 (2016)

- Multilayer structure of NbTiN = 3 nm and AlN = 2/1.5/1 nm. Series of 16-, 8-, 4-, 2-, 1- bilayers deposited on NbTiN/MgO
- Metamaterial layered superconductor of NbTiN and AlN can enhance T_c compared to NbTiN single layer.
- Low roughness of sequential films is necessary to accomplish sharp interfaces.

V. Smolyaninova

# bilayers	NbTiN [nm]	AlN [nm]
16	3	2.5
8	3.3	2.4
4	4.3	2.5
2	3.4	2

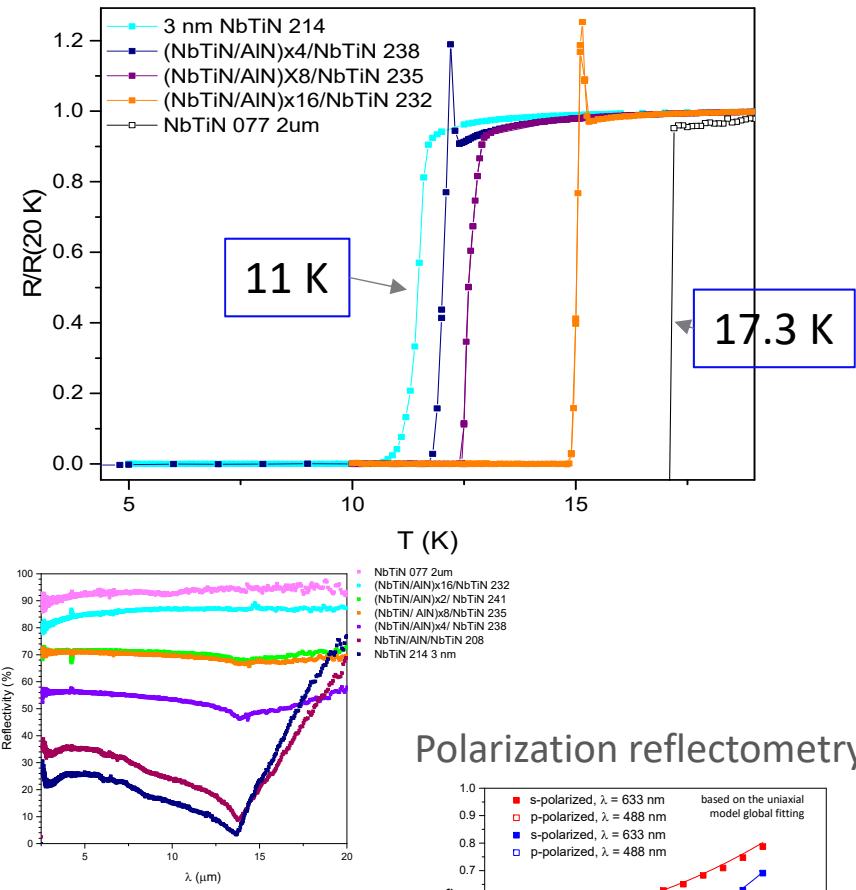
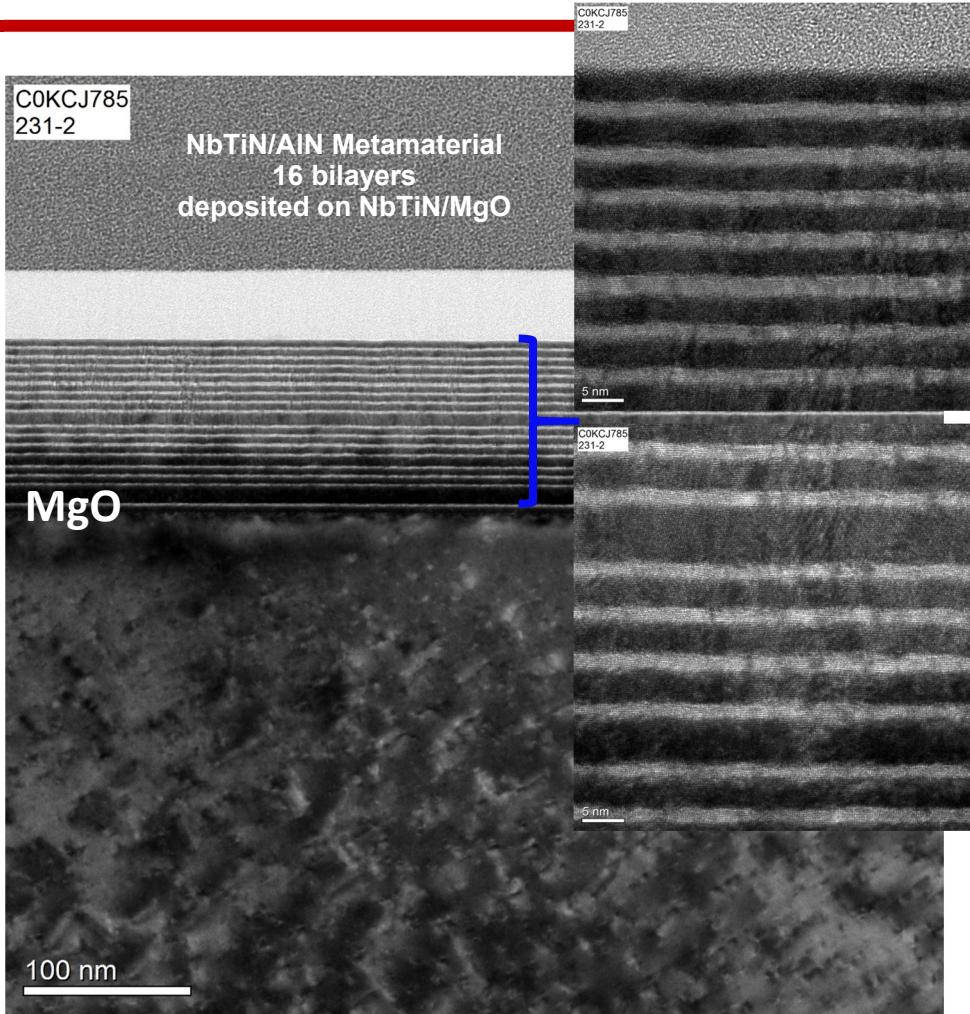
Bilayers deposited on NbTiN/MgO.



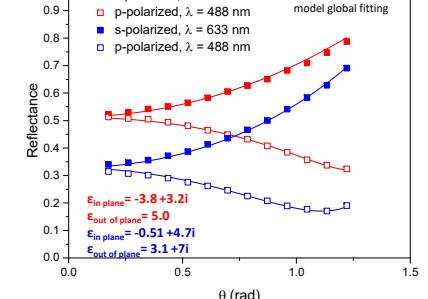
No significant increase of film roughness with additional layers.
Preserving the potential for sharp interfaces



Metamaterial superconductor based on NbTiN



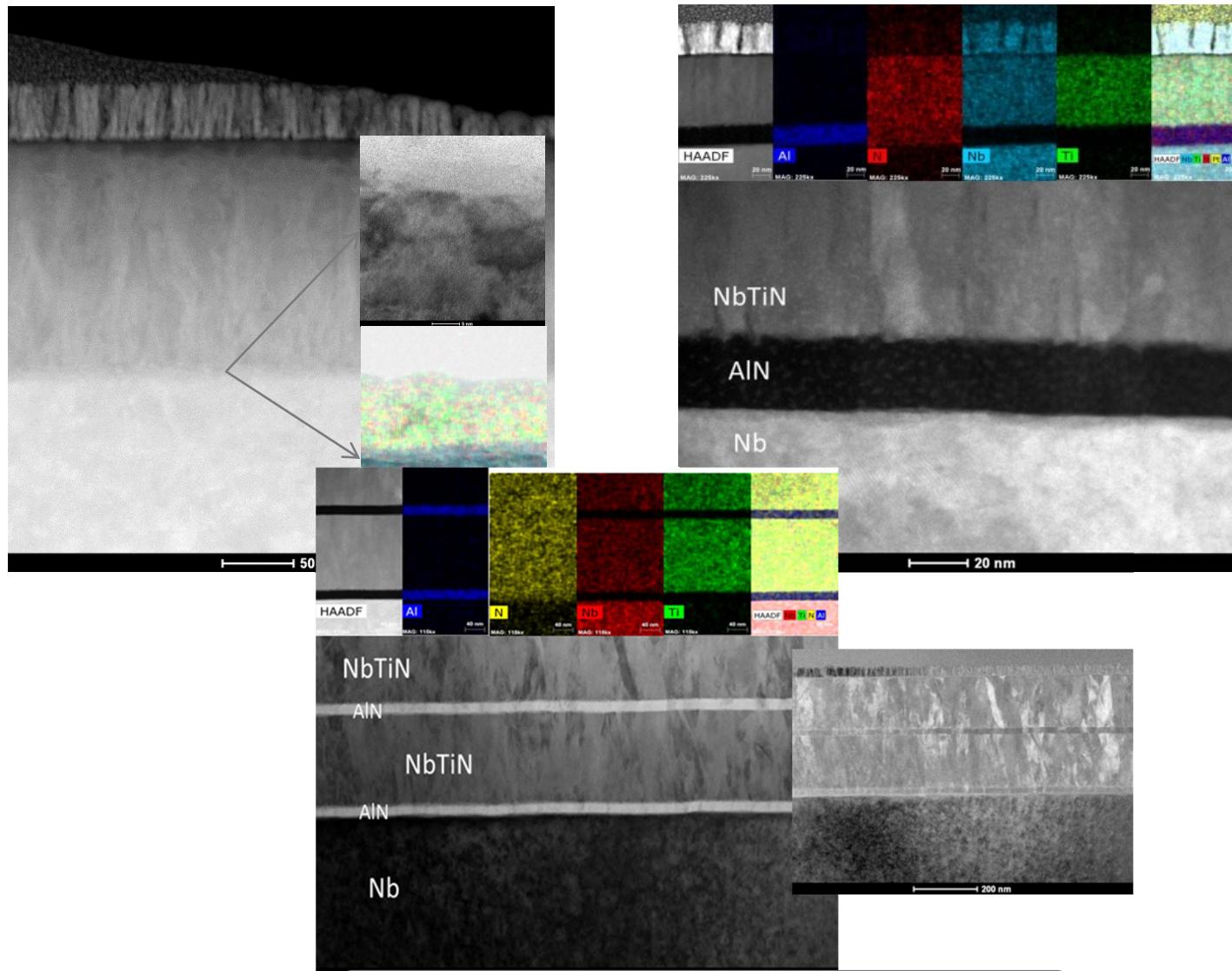
FTIR Reflectivity of
multilayers & monolayers



Hyperbolic metamaterial properties demonstrated

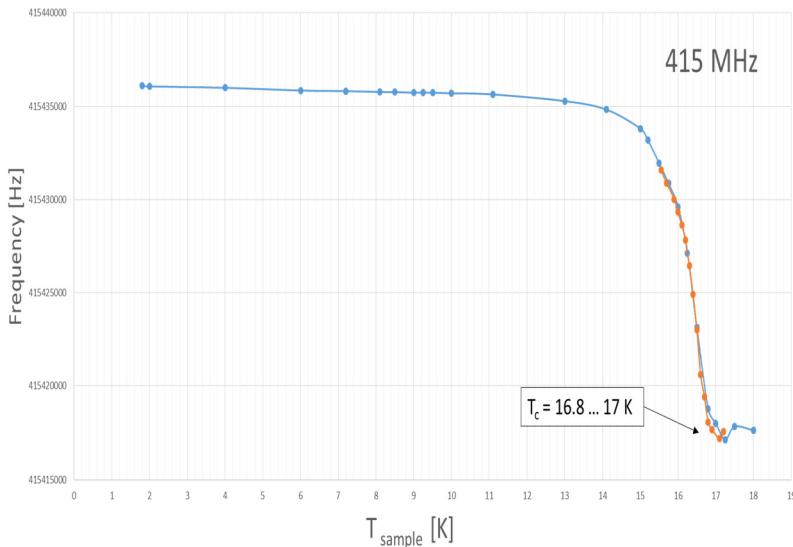
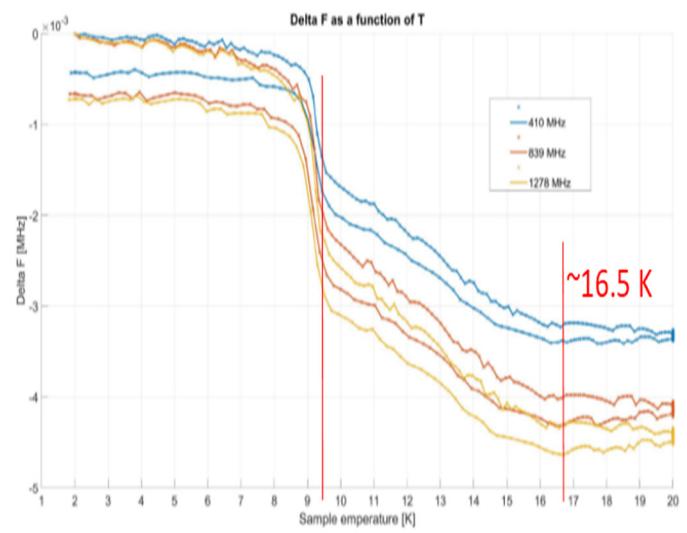
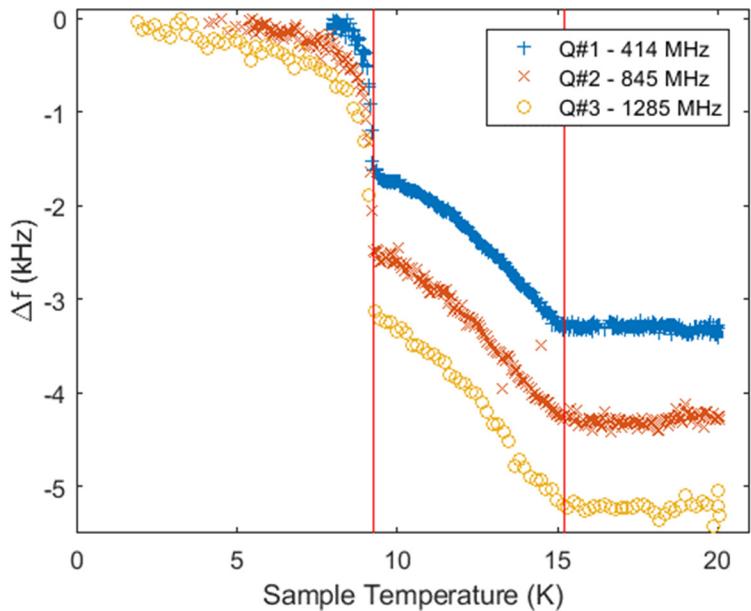
T_c enhancement with increased number of bi-layers but not yet above bulk value as predicted by theory

SIS NbTiN/AlN structures on MgO & Nb surfaces



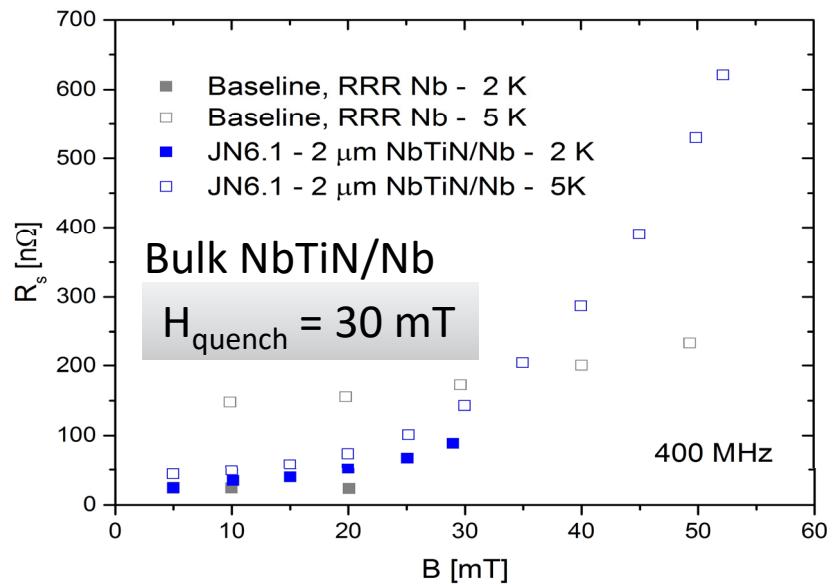
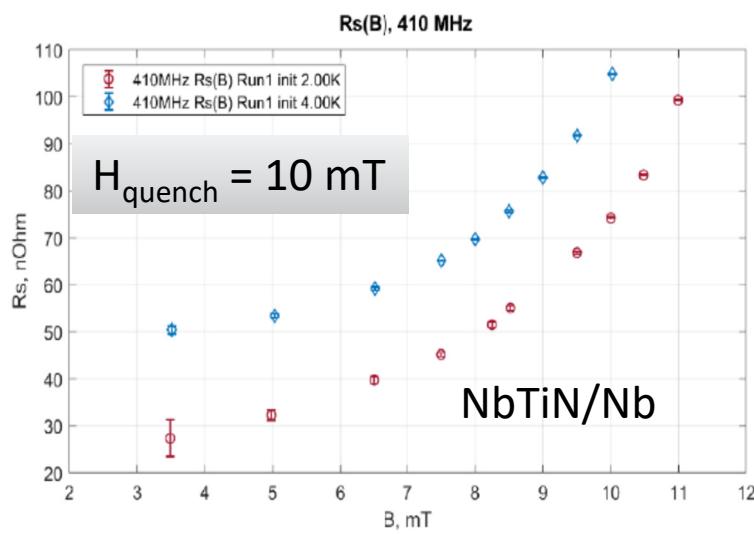
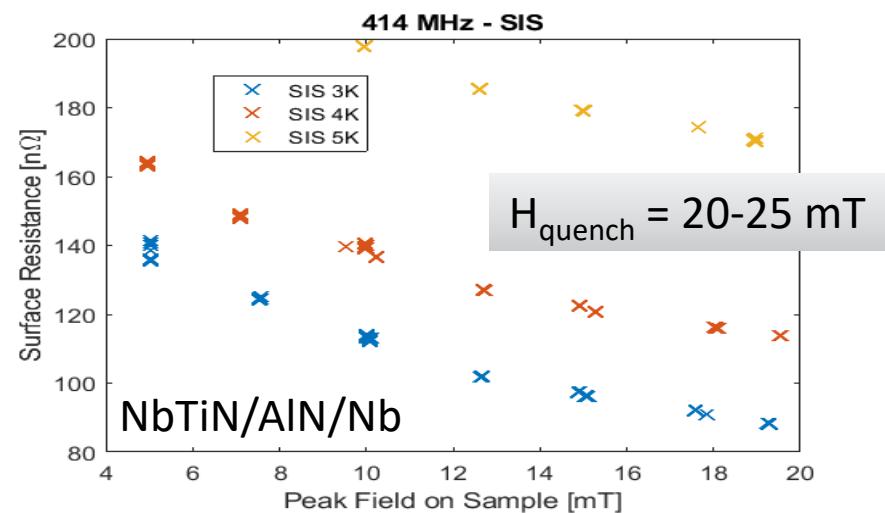
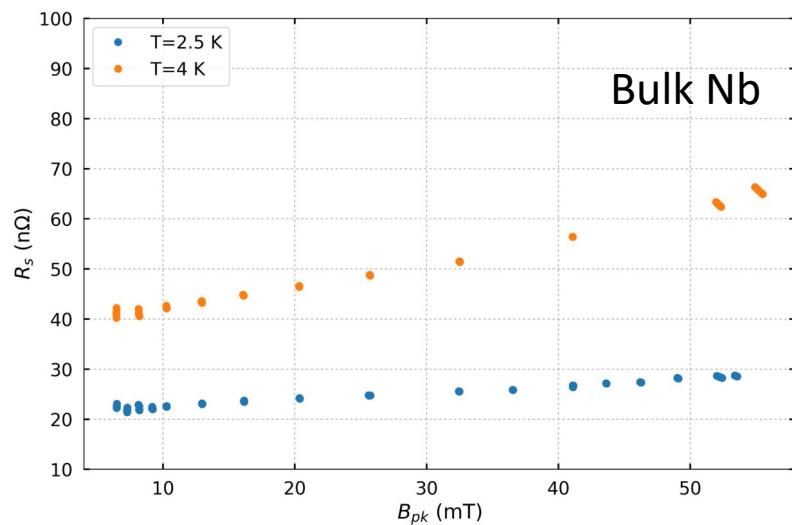
Good quality structures with sharp interfaces on Nb surfaces

NbTiN based QPR Samples – Critical Temperature

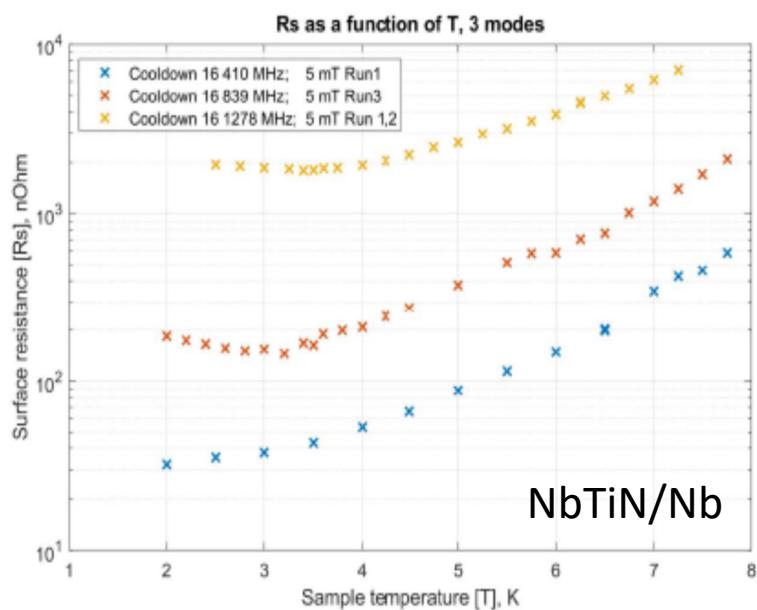
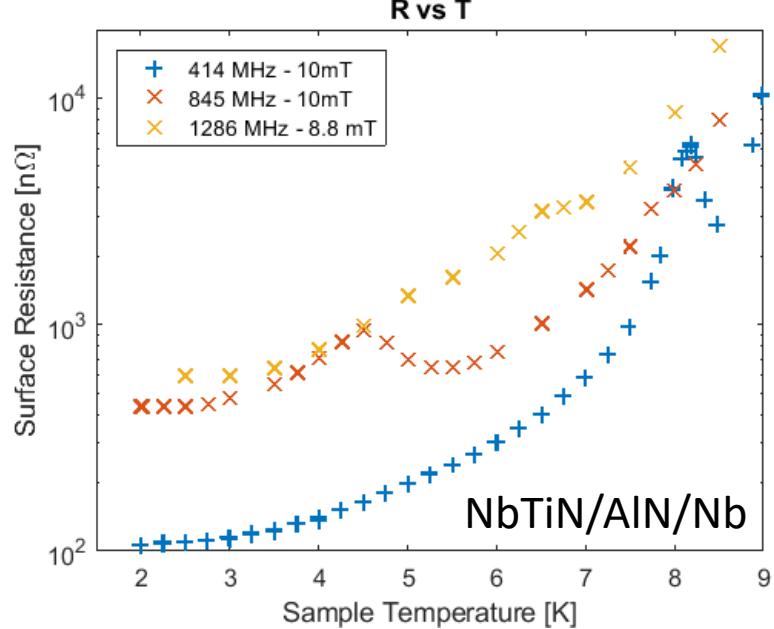
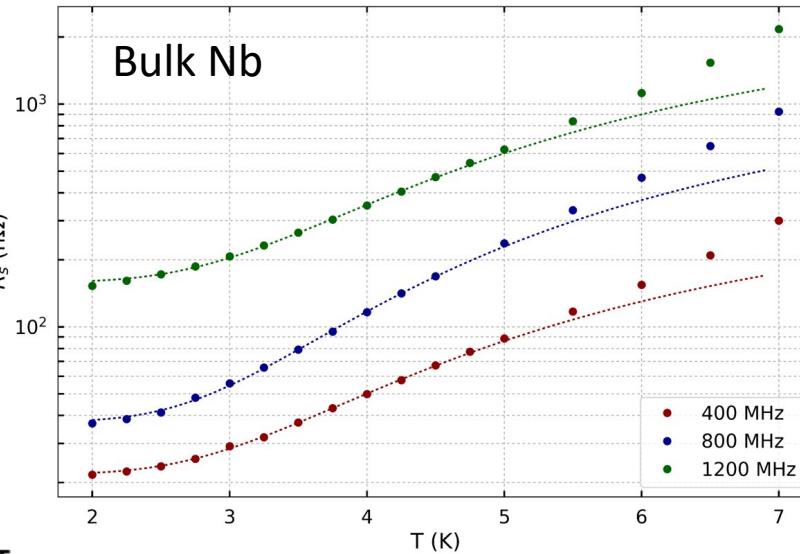


QPR measurements @ HZB
S. Keckert (THFUA1)
D. Tikhinov (TUP073)

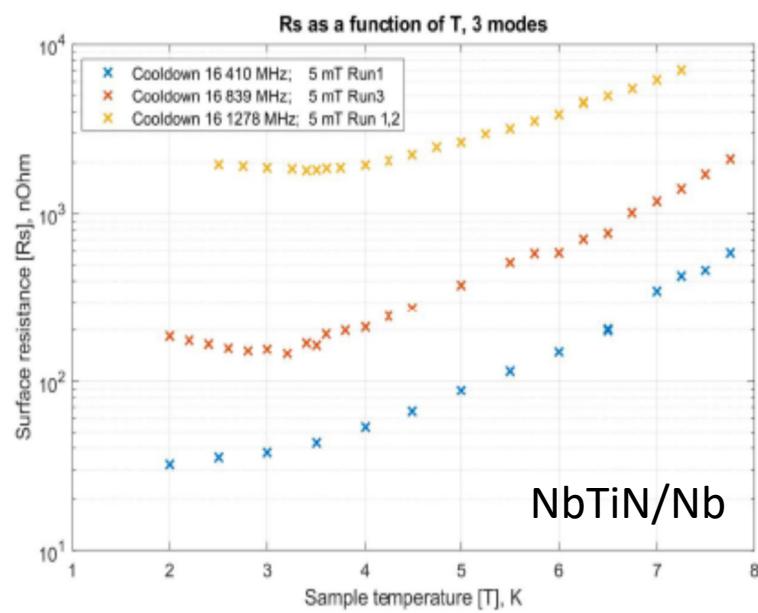
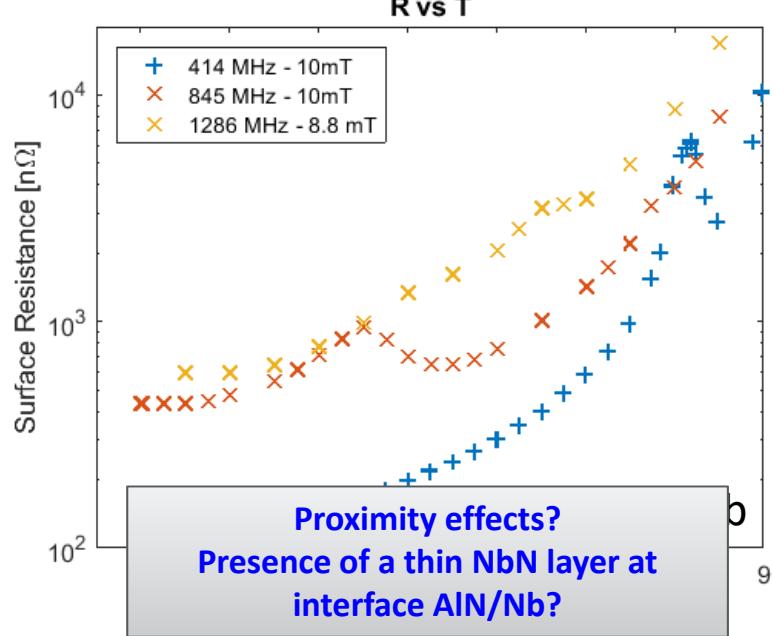
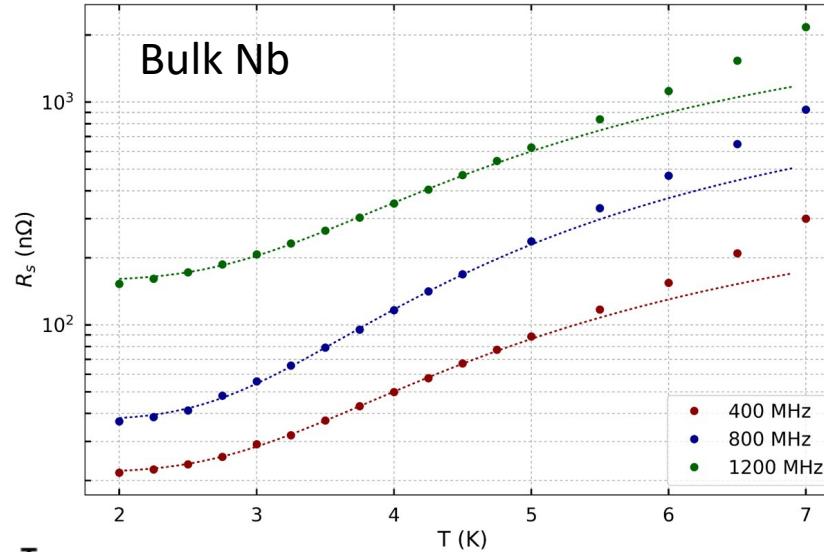
NbTiN Structures on QPR vs. Bulk Nb Baseline (415 MHz)



NbTiN Structures versus Bulk Nb Baseline – $R_s(T)$



NbTiN Structures versus Bulk Nb Baseline – $R_s(T)$



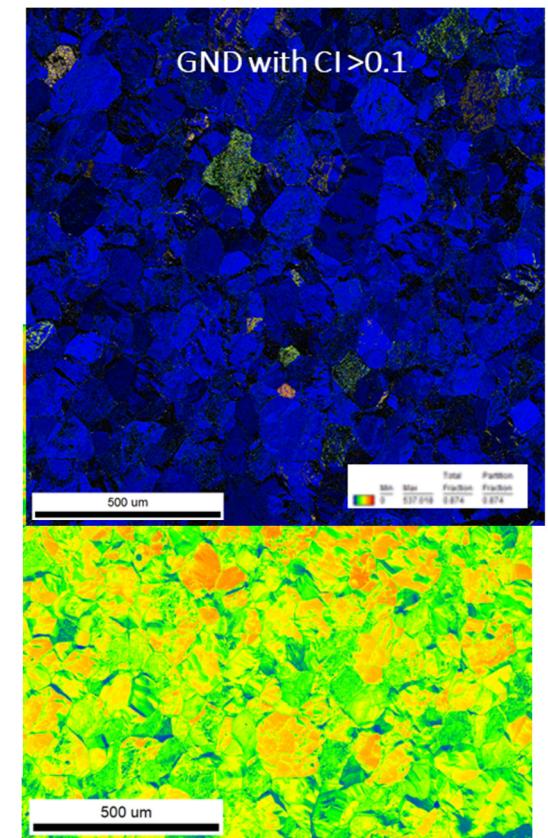
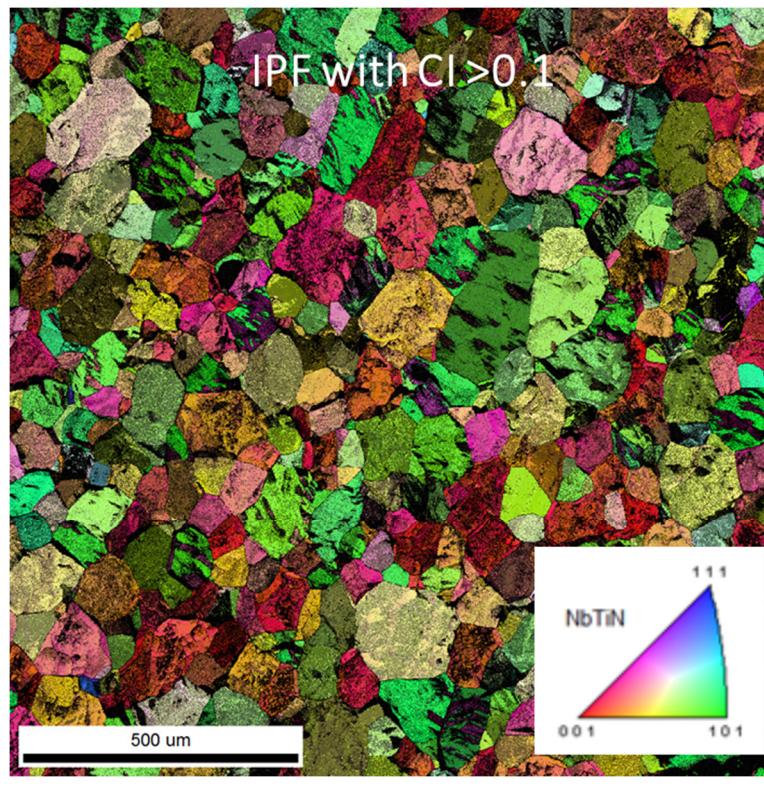
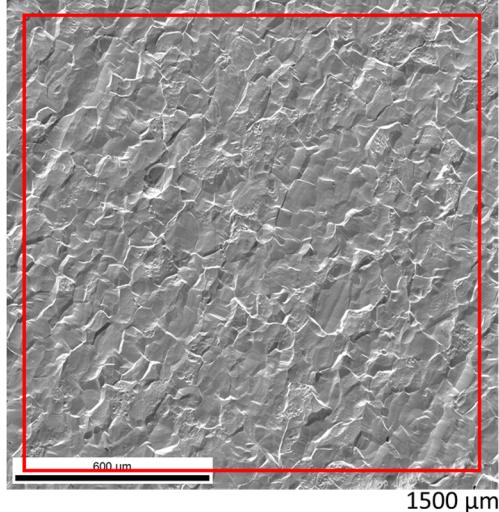
NbTiN based QPR samples structure

75 nm NbTiN/AlN/Nb

Not able to acquire EBSD map (only a few Kikuchi pattern here and there)

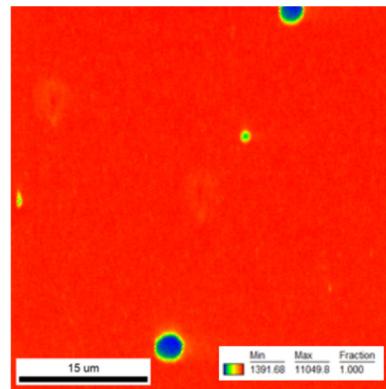
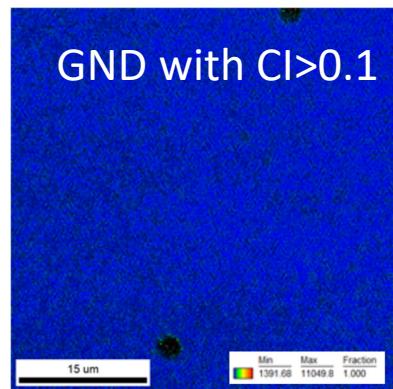
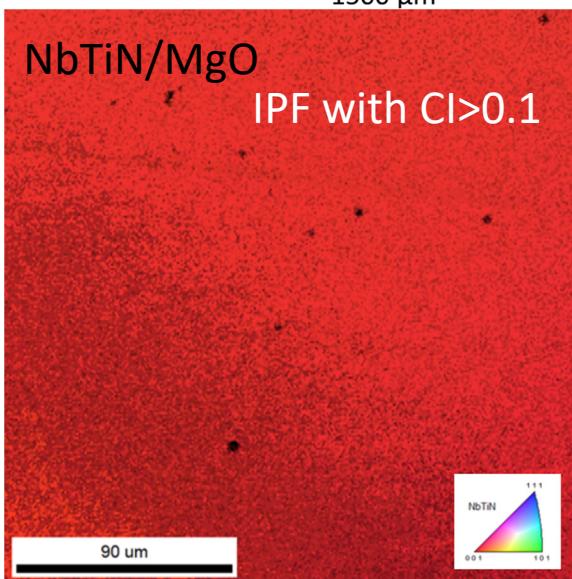
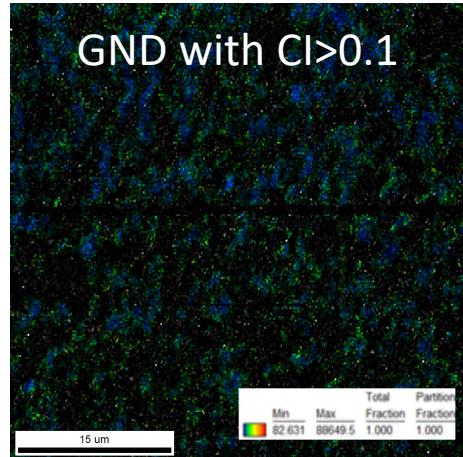
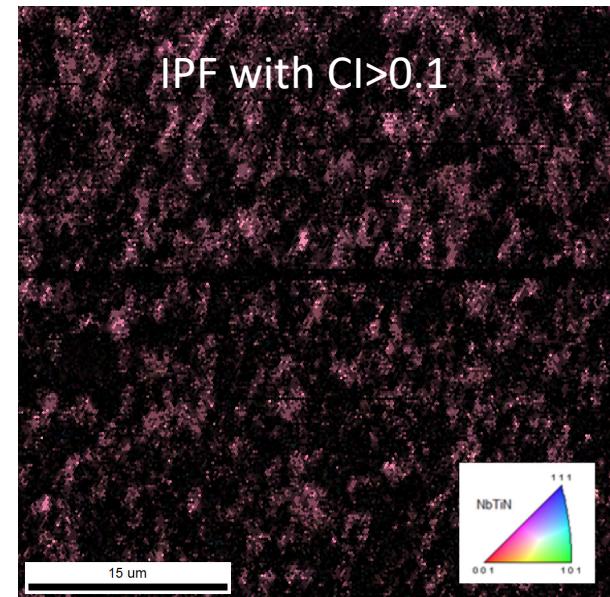
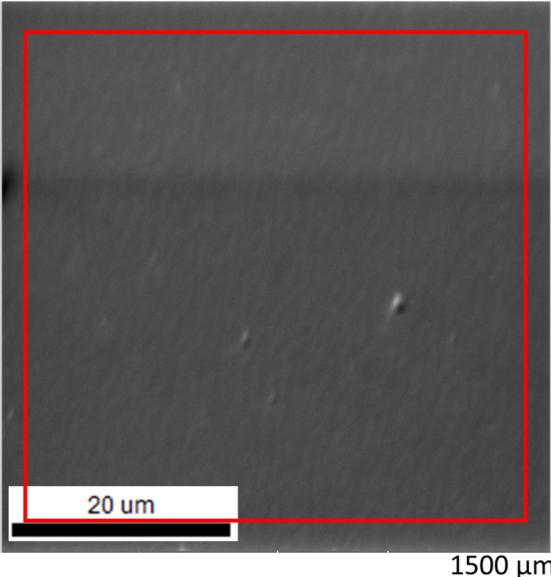
75 nm NbTiN/Nb

SE Image with EBSD Region Highlighted



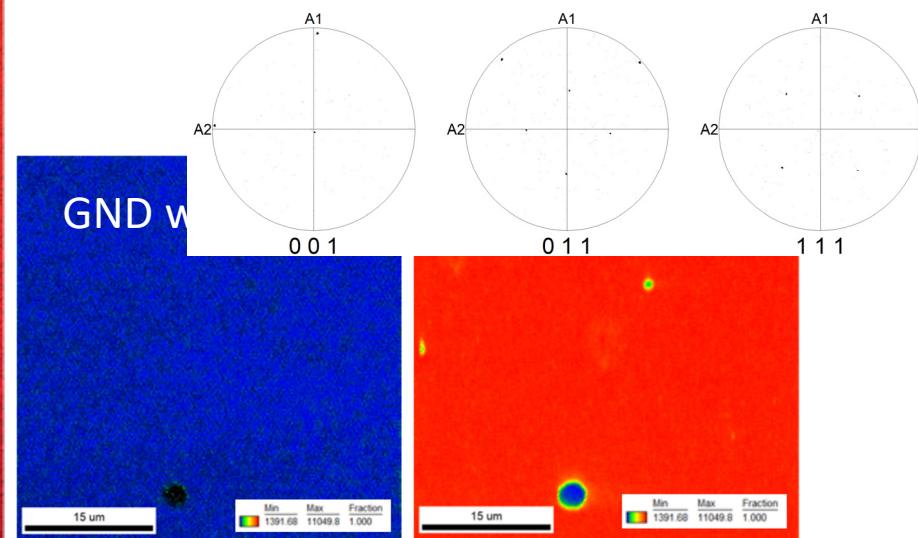
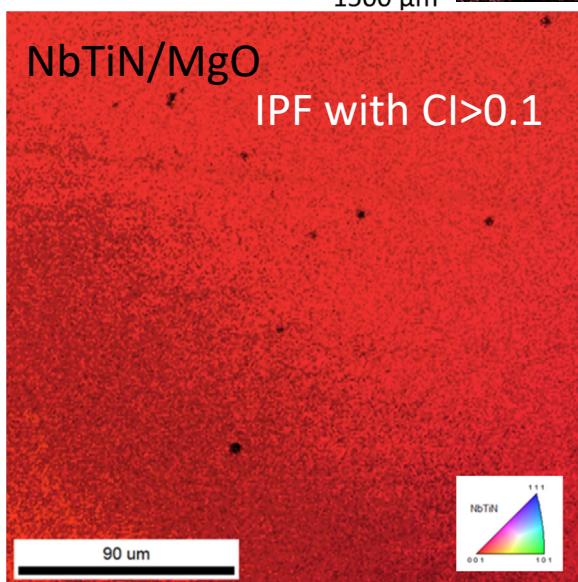
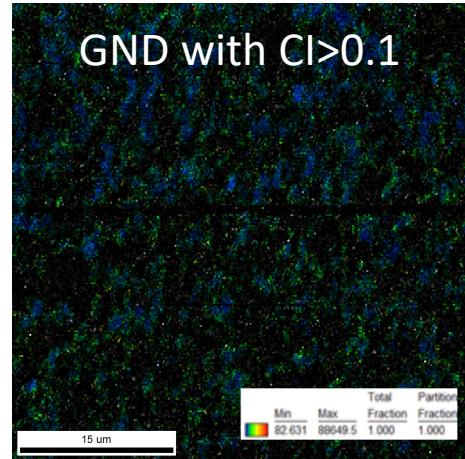
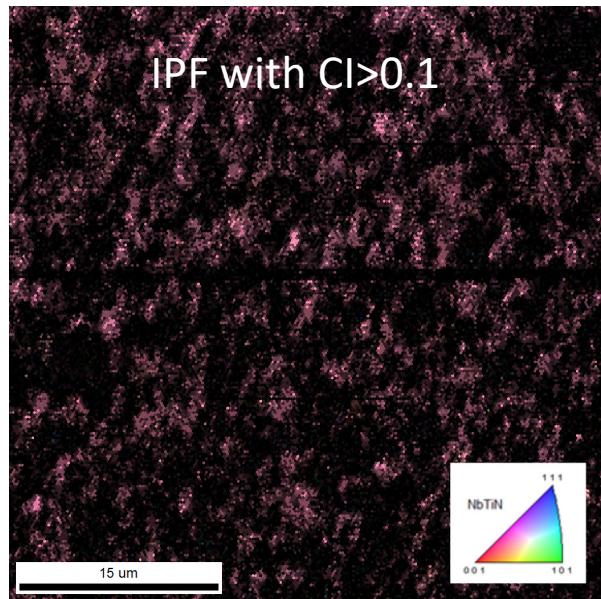
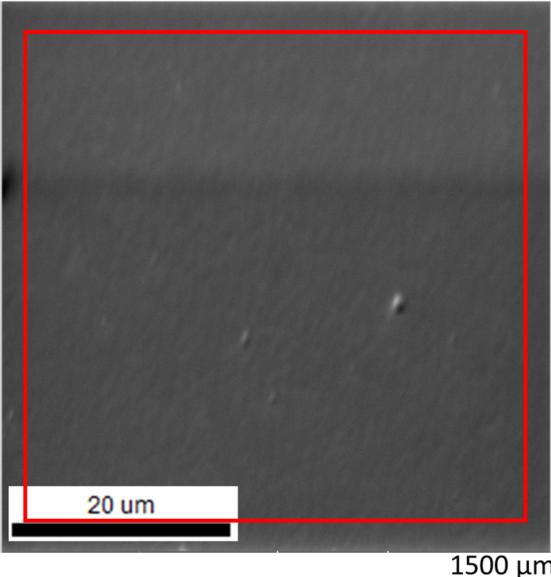
Bulk NbTiN/Nb (2 μ m)

SE Image with EBSD Region Highlighted

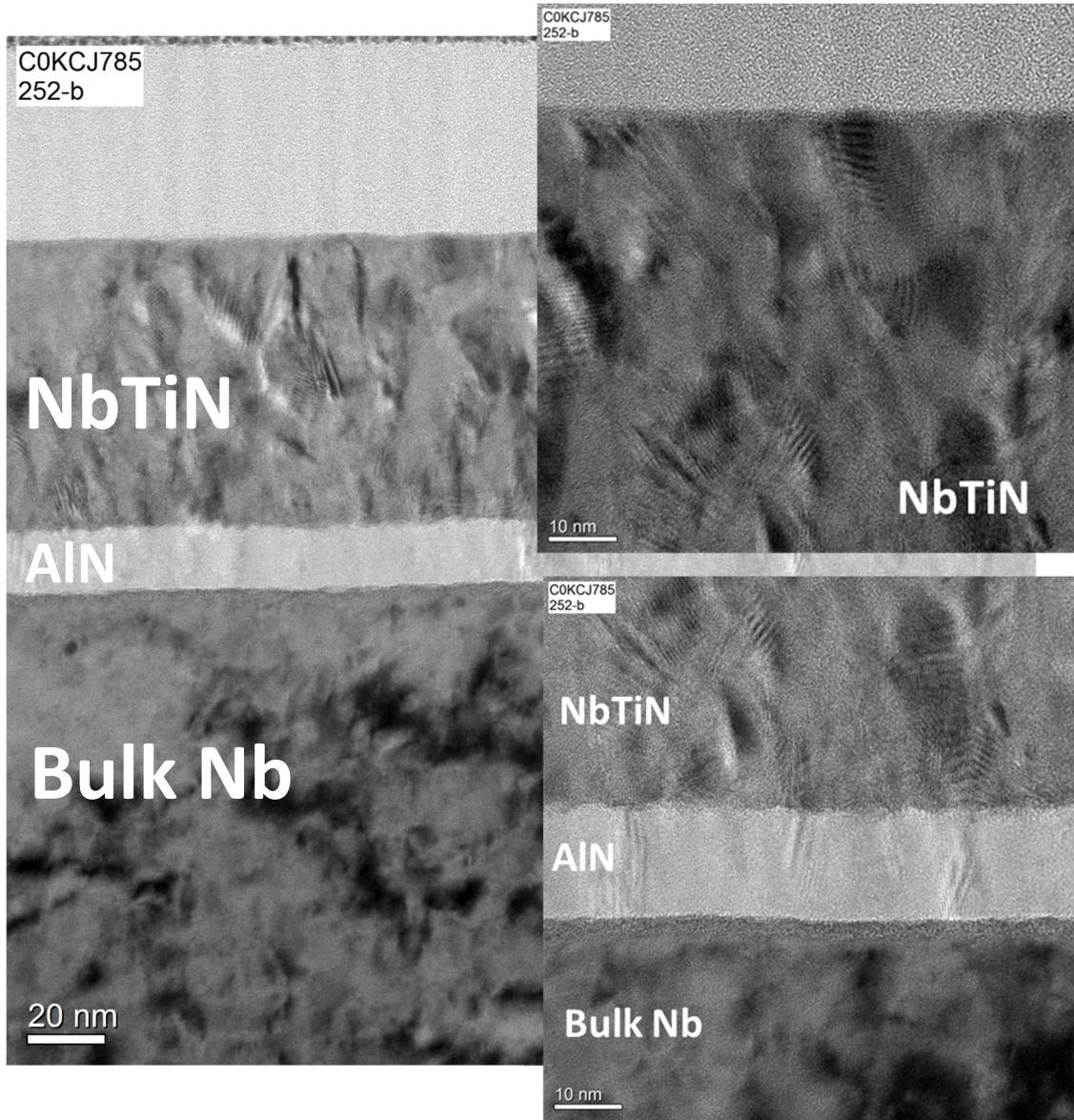


Bulk NbTiN/Nb (2 μ m)

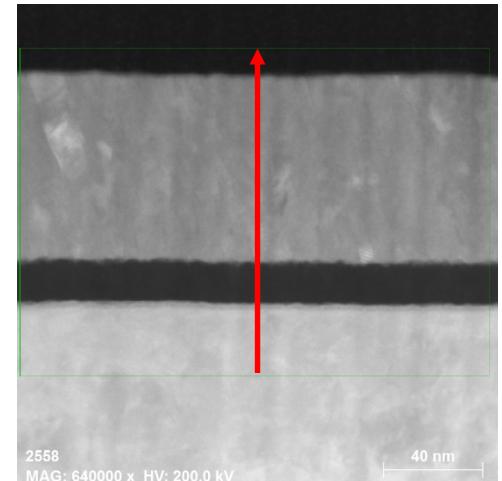
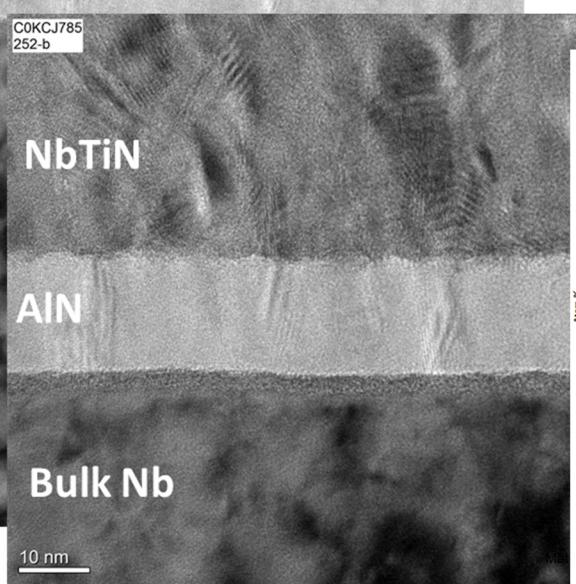
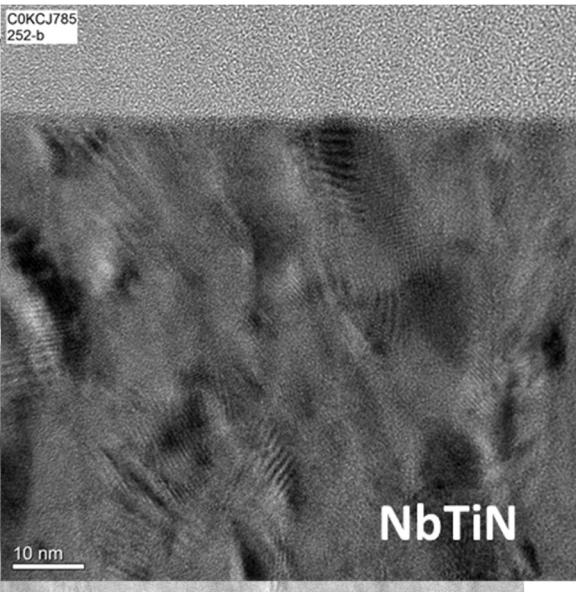
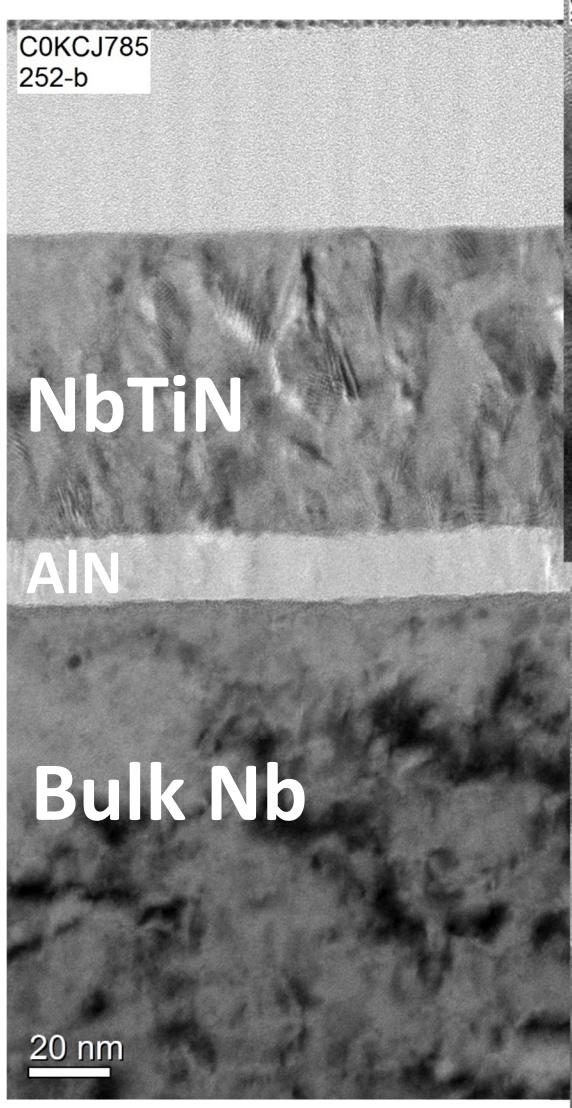
SE Image with EBSD Region Highlighted



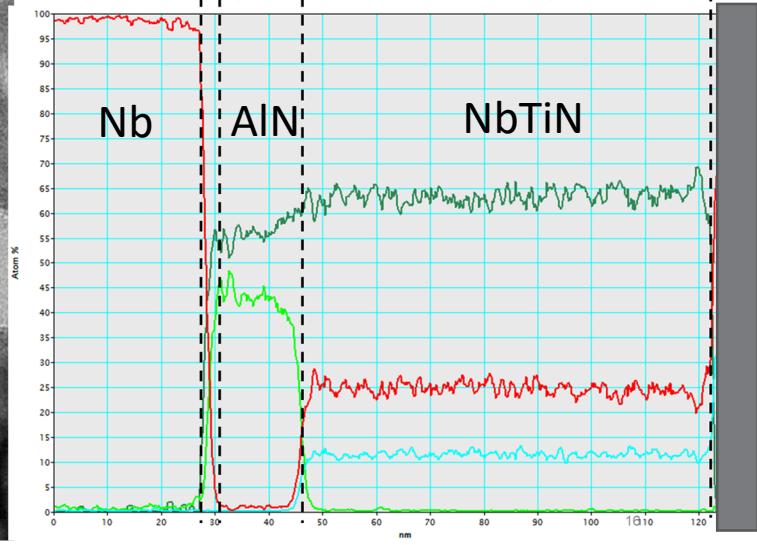
S-I-S' (75 nm NbTiN/AlN/Nb)



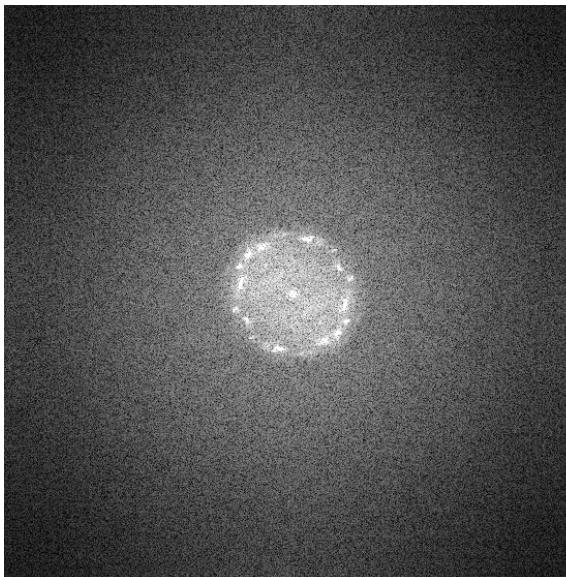
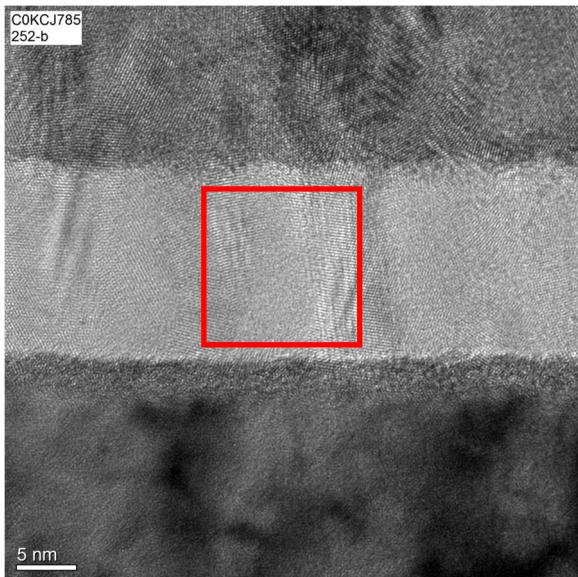
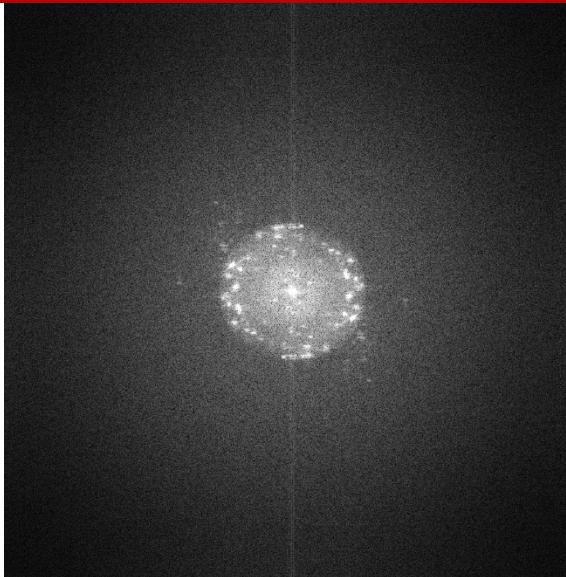
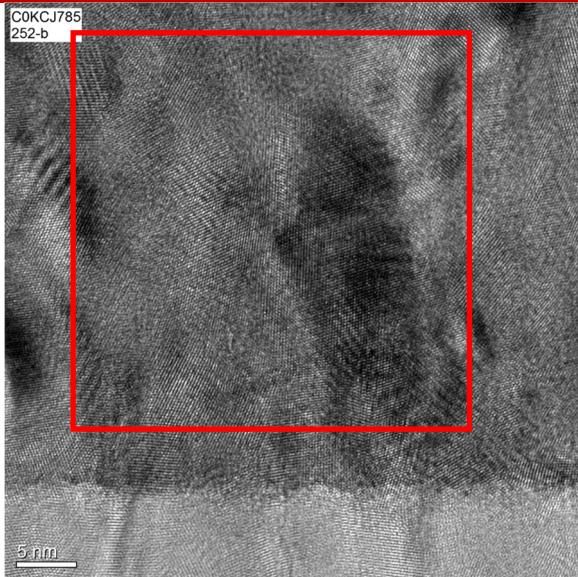
S-I-S' (75 nm NbTiN/AlN/Nb)



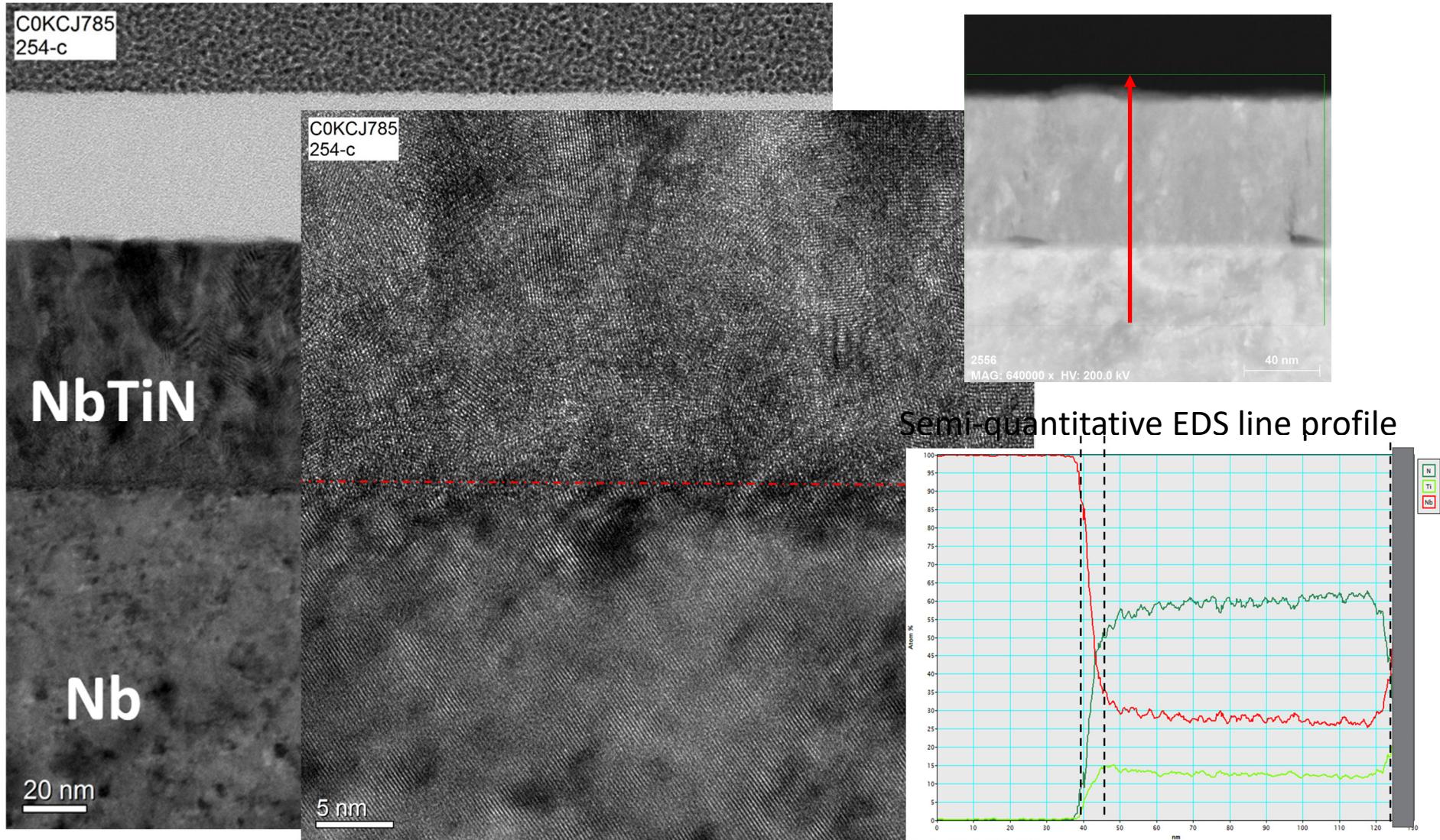
Semi-quantitative EDS line profile



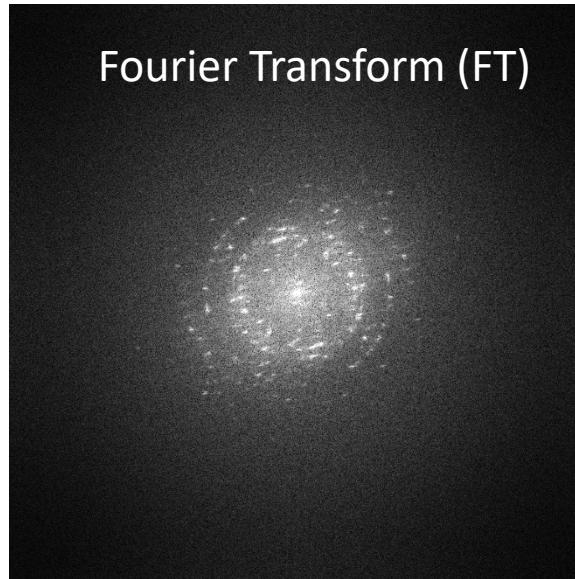
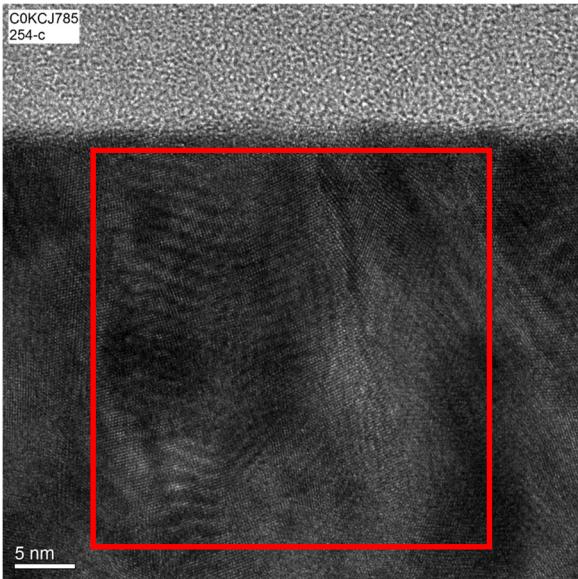
S-I-S' (75 nm NbTiN/AlN/Nb)



S-S' (75 nm NbTiN/Nb)

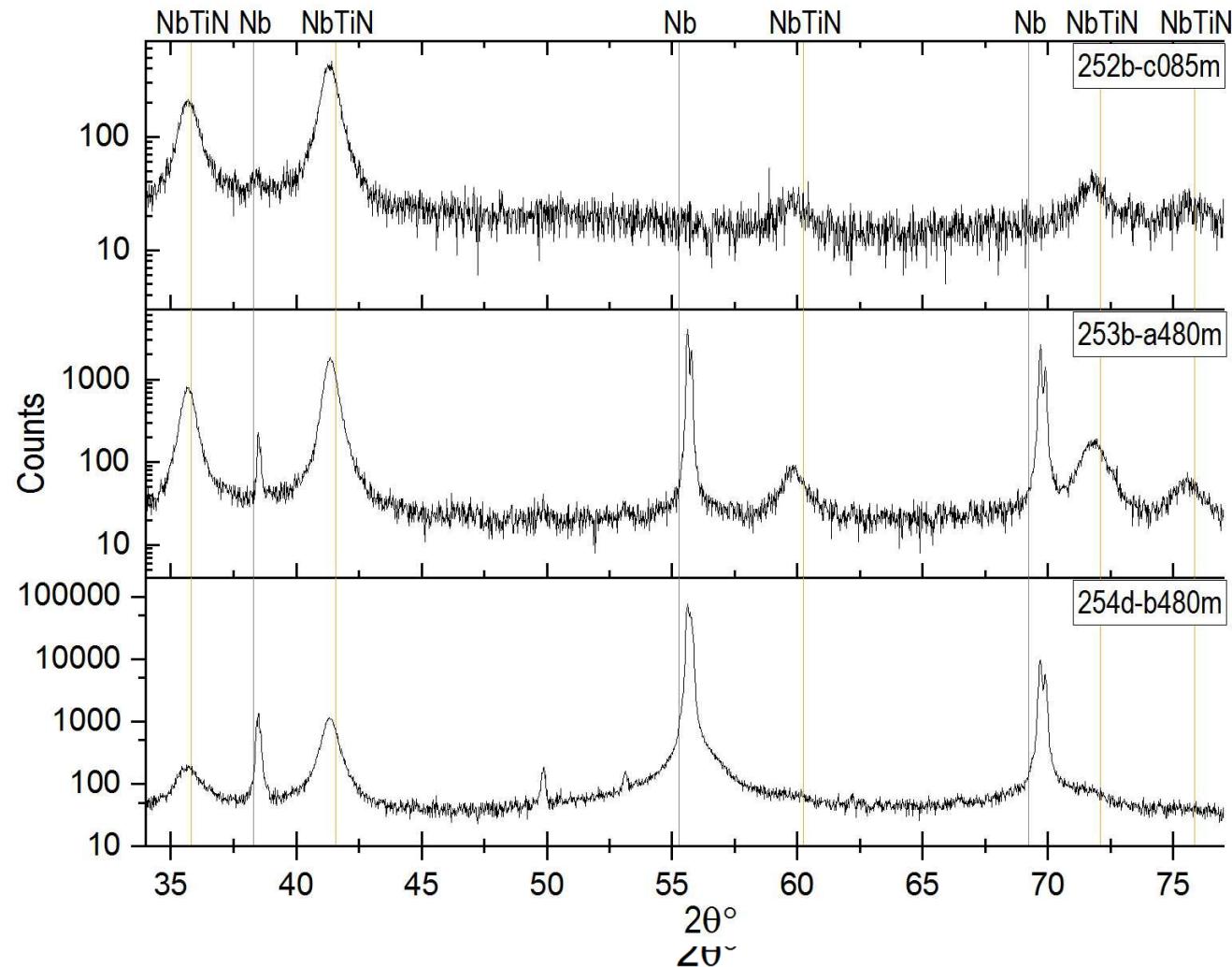


S-S' (75 nm NbTiN/Nb)

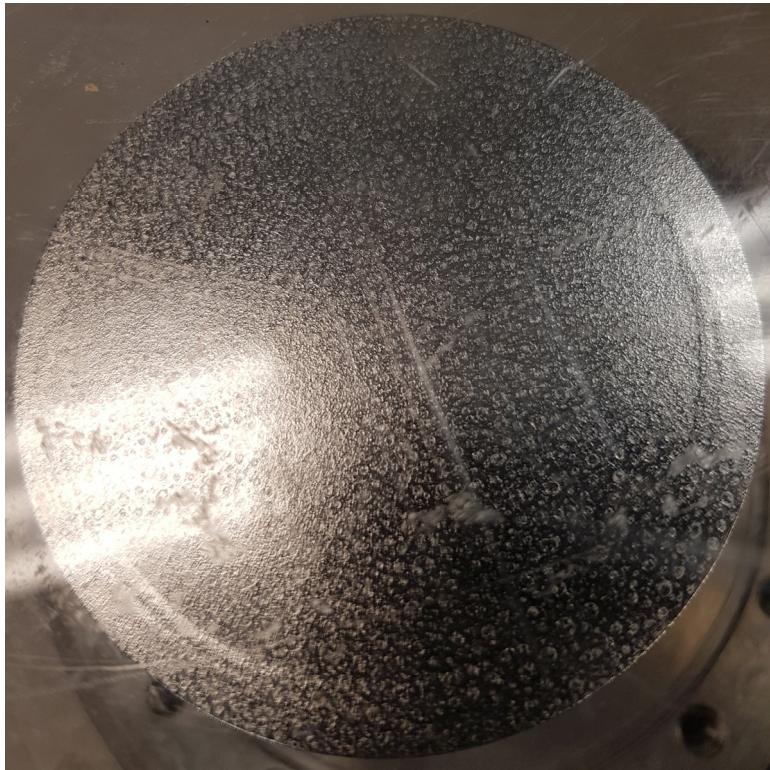


Layer is too thin for SAED. Fourier transform used to determine the crystallinity. This layer is nanocrystalline.

NbTiN based QPR samples structure



Known Issues



QPR sample after AlN removal and EP

- 1st coating cycle aborted due to short in NbTi Magnetron source
- AlN removal with KOH
(Need to establish new chemistry for alternative materials)
AlN residues still present?
- Features appeared on the Nb surface, only removed with extensive EP

Potential H_{\max} for NbTiN

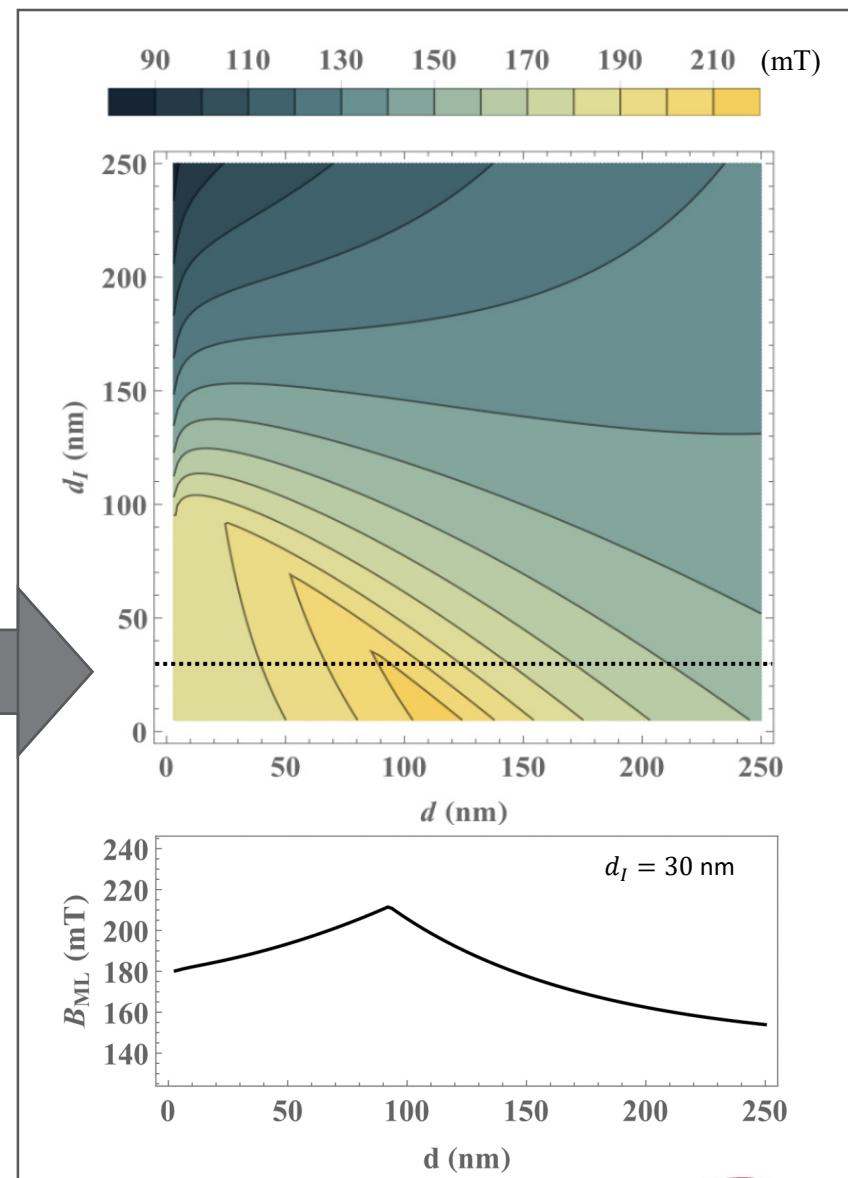
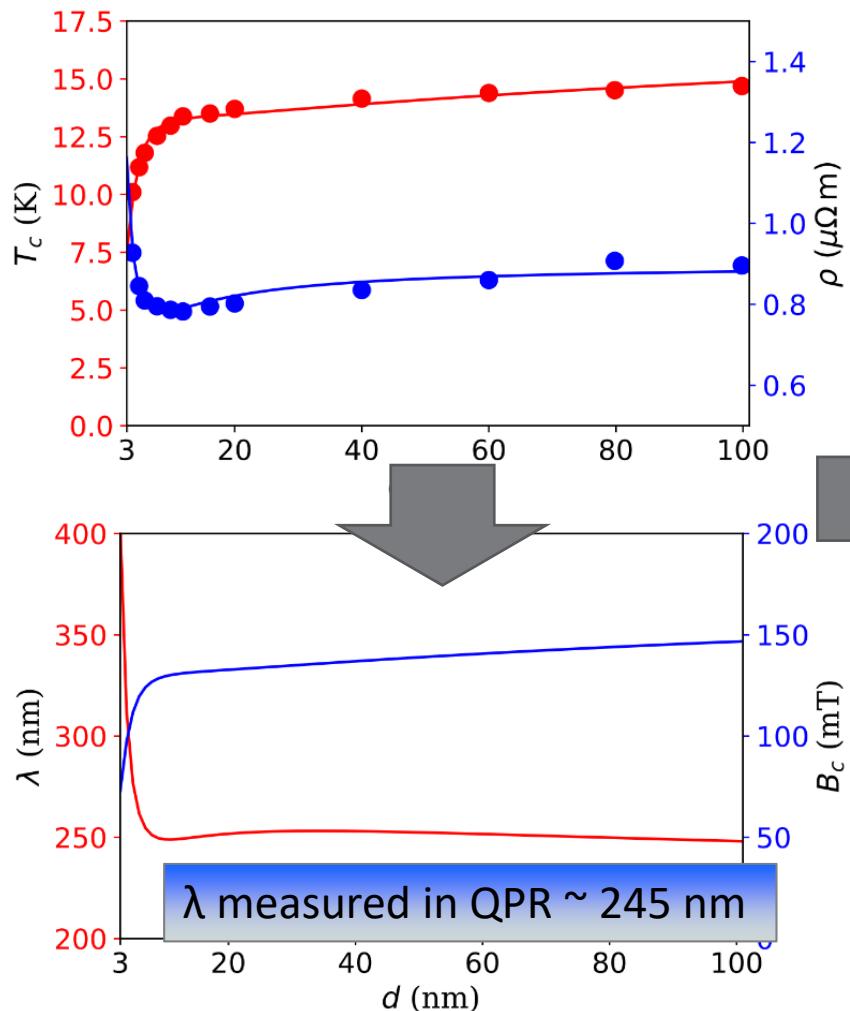
Courtesy T. KUBO

T_c and ρ from L. Zhang et al., Appl. Phys. Lett. 107, 122603 (2015)

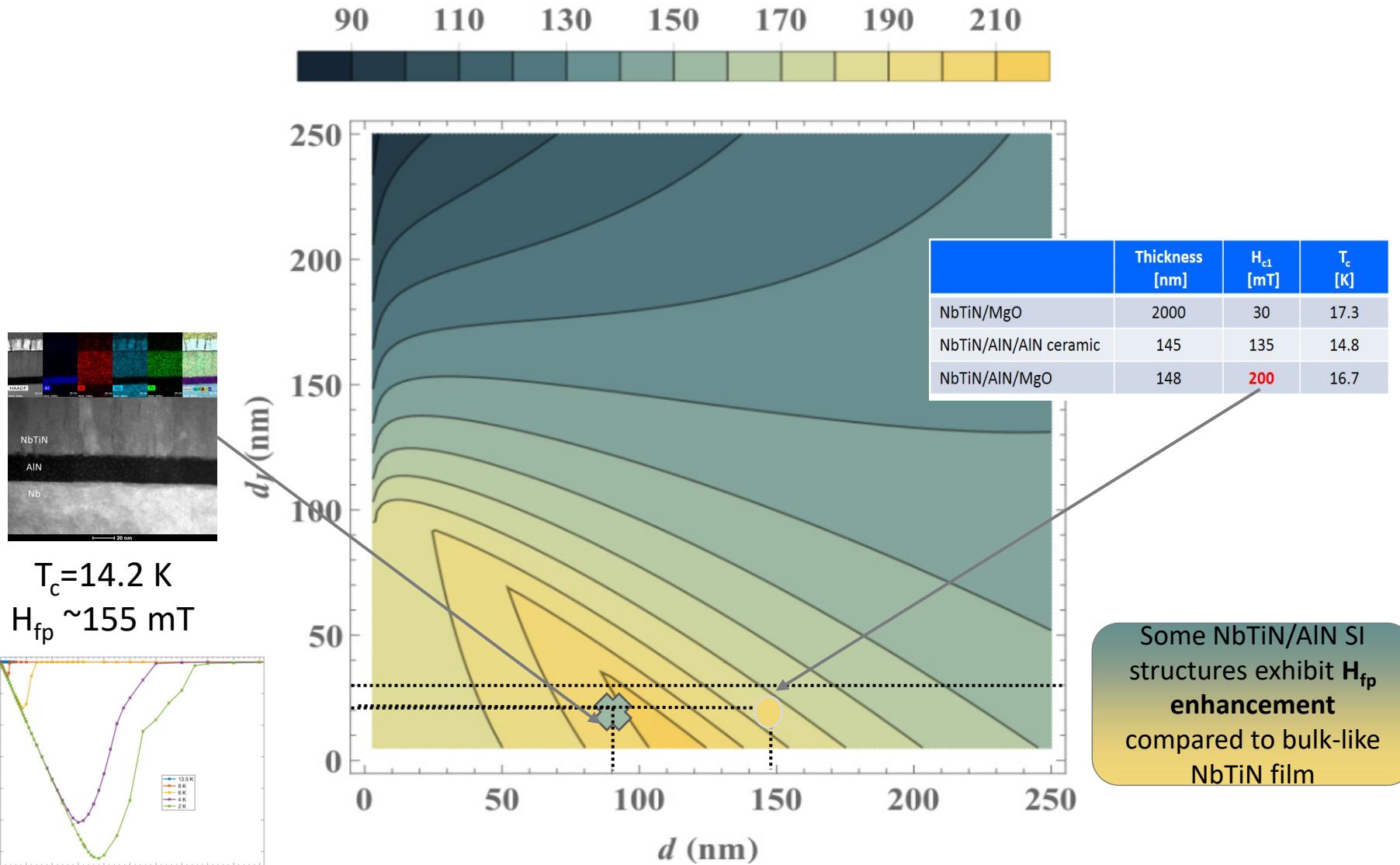
$\Delta = 1.86 k_B T_C$ from T. Hong et al., J. Appl. Phys. 114, 243905 (2013).

$N_0 = 1.17 \times 10^{47} J^{-1} m^{-3}$

from D. Hazra et al., Phys. Rev. B 97, 144518 (2018)



Potential H_{\max} for NbTiN



CONCLUSION & FUTURE WORK

In depth analysis of RF behavior ongoing across frequencies, cooling and applied magnetic field condition

**3rd harmonic measurements for first penetration field measurements ongoing
(CEA Saclay, KEK)**

- Thickness and T_c on large sample are consistent with small sample studies
 - Quench field are limited to lower values than predicted
 - Films and structures follow substrate topology

- Refine the contour plot for actual films coated
- Repeat the S-I-S' on QPR sample with no surface weld
- Continue exploring the NbTiN/AlN thickness with QPR and 3rd harmonic measurements
- Reactive HiPIMS to densify the films and get relaxed structure at earlier growth stage
- Explore other superconductors matched with adequate dielectric

