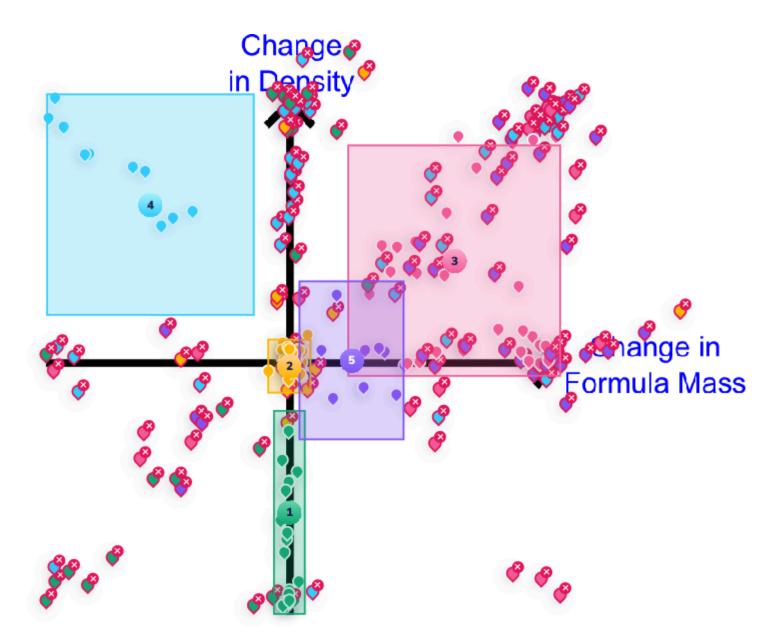
# Lecture 3 - Ionic Conductivity

### Lecture summary

- Recap of defect types
- Ionic conductivity
- Conduction mechanisms
- Ionic migration paths
- Energetics of conduction

## Defect recap

## Defect recap results



#### **NaCl Schottky**

- Intrinsic defect (composition doesn't change) so formula mass change is 0
- Vacancies essentially create a hole in the structure, so density decreases

#### **AgCI Frenkel**

- Intrinsic defect
- Vacancies and interstitials created at the same rate, so density doesn't change significantly

#### TiO2 substitution with Zr

- Zr is heavier than Ti (91.224 vs 47.867) so formula mass will increase
- Zr<sup>4+</sup> is slightly larger than Ti<sup>4+</sup> so we might expect volume to increase slightly, but overall the density will increase
  - NOTE: I wrongly said Zr<sup>4+</sup> was smaller than Ti<sup>4+</sup> in the lecture sorry!

#### Shear phase in WO<sub>3</sub>

- ullet  $\mathrm{WO_3} \longrightarrow \mathrm{WO_{3-v}}$ , so the formula mass will decrease
- Shear phase formation is a substantial rearrangement which is driven by reducing the overall volume.
   The result is that decrease in volume far outweighs the decrease in formula mass, so density will increase.

#### Replacing O by F in CeO<sub>2</sub>

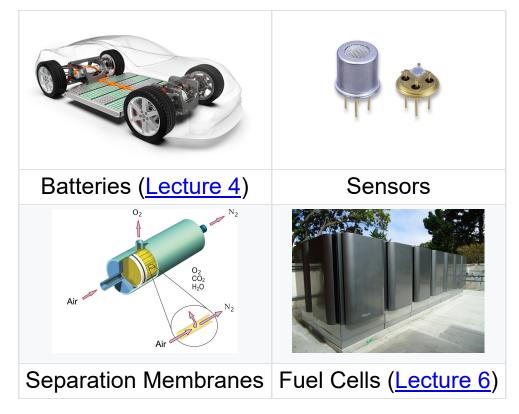
- $CeO_2 \longrightarrow CeO_{2-x}F_x$
- F is slightly heavier than O (18.998 vs 15.999) so formula mass will increase slightly
- Charge balance requires us to reduce some Ce<sup>4+</sup> to Ce<sup>3+</sup>. There are a few ways to imagine this with KV notation:
  - 1. Replacing  $O^{2-}$  by  $F^{1-}$  occurs directly during synthesis:
    - $\bullet \ {\rm O_O} + {\rm Ce_{Ce}} + {\rm H_2} + \tfrac{1}{2} {\rm F_2} \longrightarrow {\rm F_O^{\bullet}} + {\rm Ce_{Ce}^{'}} + {\rm H_2O}$
    - (H<sub>2</sub> and F<sub>2</sub> are used to balance the equation here, but a real experiment would use a different fluorinating reagent such as CeF<sub>3</sub>)
  - 2. As-synthesised CeO<sub>2</sub> is fluorinated by filling existing oxygen vacancies:
    - $V_{O}^{\bullet \bullet} + Ce_{Ce}^{'} + \frac{1}{2}F_{2} \longrightarrow F_{O}^{\bullet} + Ce_{Ce}^{x}$
    - Note that the fluorite structure is often prone to anion vacancies due to the mismatch between ionic radii and the geometry (this is particularly prevalent when the cation can be reduced). We'll see why this again in lecture 6.

### Conductivity

- Many ionic solids conduct electricity; due to ionic and/or electronic motion.
- Most ionic solids are electrically insulating/semiconducting (localised electrons)

#### Conductivity

- Many ionic solids conduct electricity; due to ionic and/or electronic motion.
- Most ionic solids are electrically insulating/semiconducting (localised electrons)
- Ionic conductors are important!



## Origin of ionic conduction

- Ionic conductivity is dominated by defects
  - In an ideal crystal, ions can't easily move
  - vacancies and/or interstitials are the main charge carriers

### Origin of ionic conduction

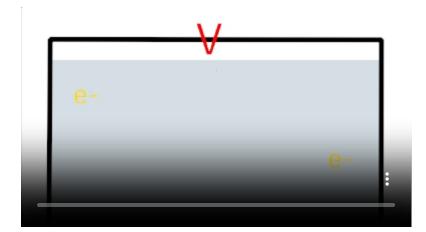
- Ionic conductivity is dominated by defects
  - In an ideal crystal, ions can't easily move
  - vacancies and/or interstitials are the main charge carriers
- Conductivity,  $\sigma=nq\mu$ , where
  - ∘ *n* is number of charge carriers
  - ∘ *q* is charge
  - $\circ \ \mu$  is the mobility of charge carriers

## Origin of ionic conduction

- Ionic conductivity is dominated by defects
  - In an ideal crystal, ions can't easily move
  - vacancies and/or interstitials are the main charge carriers
- Conductivity,  $\sigma = nq\mu$ , where
  - n is number of charge carriers
  - ∘ *q* is charge
  - $\circ \mu$  is the mobility of charge carriers
- In ionic solids, conductivity covers  $10^{\,-16}~\mathrm{S~m^{-1}}$   $-10^{\,3}~\mathrm{S~m^{-1}}$ 
  - $\circ$  most solids are limited to around  $10^{\,-2}~\mathrm{S}~\mathrm{m}^{-1}$
  - $\circ$  Liquid electrolytes typically  $10^{\,-1} 10^{\,3} \; \mathrm{S} \; \mathrm{m}^{-1}$

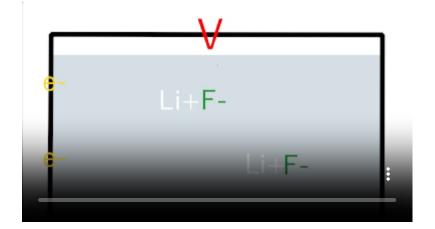
## Measuring Conductivity

- For electronic conductors, this is simple:
  - $\circ$  Apply a voltage (V) and measure the resulting current (I)
  - $\circ$  Resistance (in  $\Omega$ ) is found through Ohm's law; V = IR
  - $\circ$  Resistivity (in  $\Omega$  cm) of the material calculated from geometry
- Resistivity  $\rho$  (in  $\Omega$  cm) =  $\frac{1}{\text{Conductivity } \sigma \text{ (in S cm}^{-1)}}$



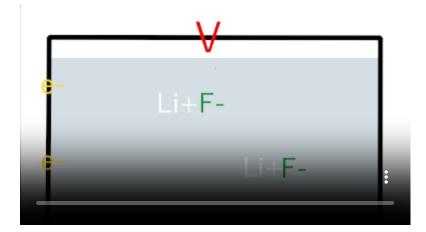
### Measuring Ionic Conductivity

• Ions cannot flow round a circuit, so current drops with a constant applied voltage

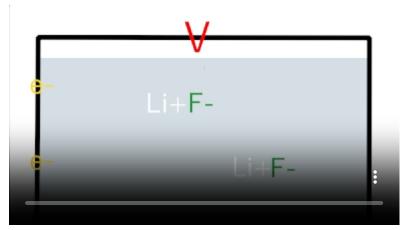


#### Measuring Ionic Conductivity

• Ions cannot flow round a circuit, so current drops with a constant applied voltage



• Instead, we use an alternating voltage - this is called Impedance spectroscopy (see lecture 5)

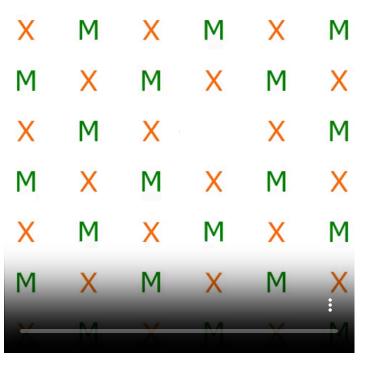


#### Ion migration mechanisms

Three 'main' mechanisms of ionic migration

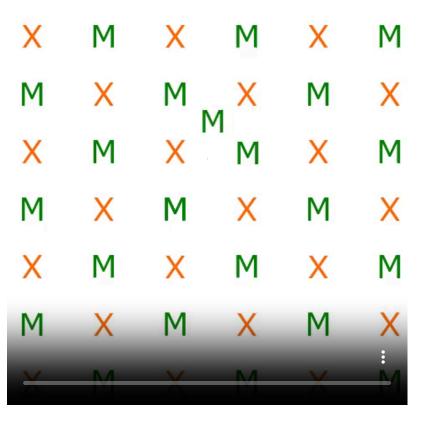
#### 1. Vacancy mechanism

Vacancies move throughout the lattice (atoms move into vacancy)



### 2. Interstitial mechanism

lons hop between interstitial sites



## 3. Interstitialcy (knock-on) mechanism

Interstitial ions 'push' into a neighbouring site



Vacancy, Interstitial or Interstitialcy?

## Suggestions

What technique(s) could you use to distinguish the mechanism?

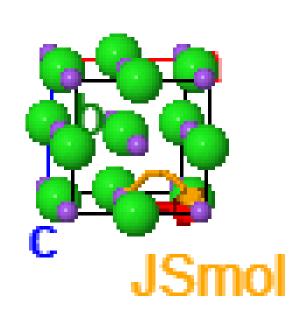
31 respondents

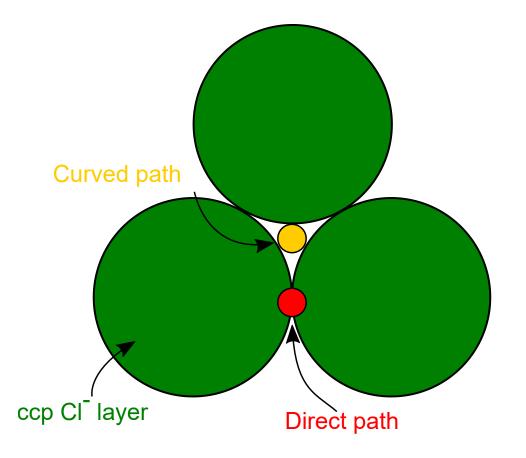
molecular dynamics simulation



## Migration paths

Ion paths are rarely direct, but will take the lowest energy route.

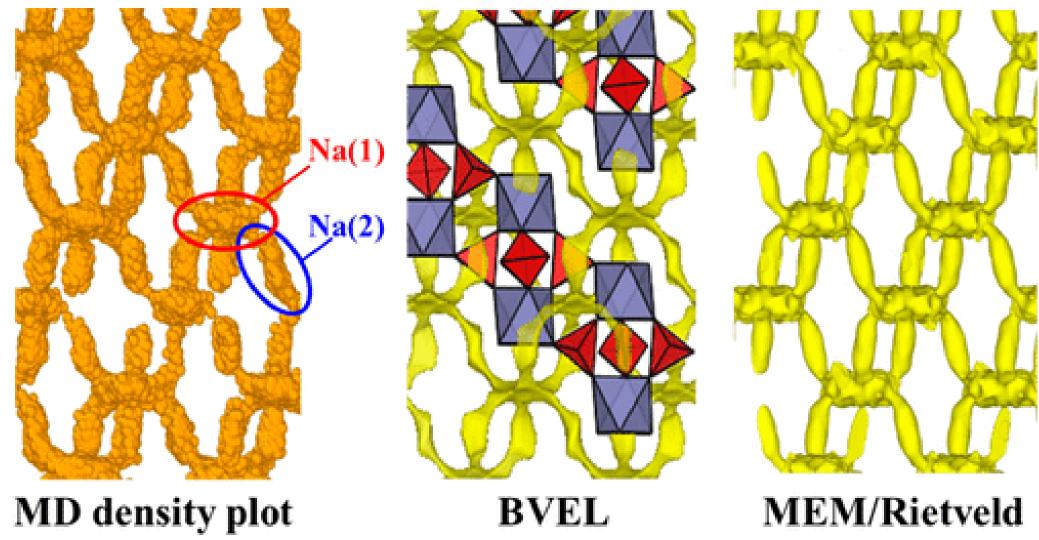




#### Pathways can be complex

Migration pathways can be calculated and/or experimentally determined

e.g. NASICON  $\mathrm{Na^{+}}$  conductor,  $\mathrm{Na_{3}Zr_{2}(SiO_{4})_{2}(PO_{4})}$ :



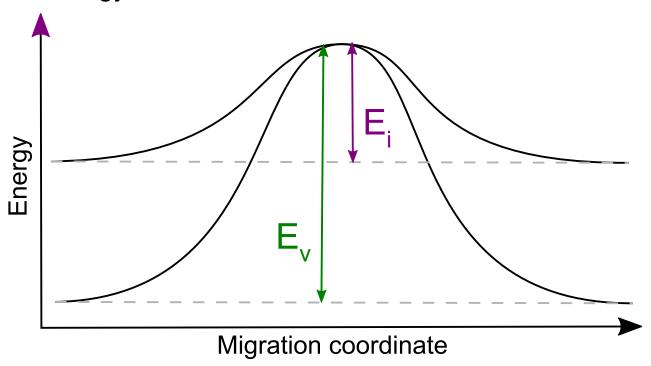
Y. Deng, Chem. Mater., 2018, 2618.

#### Migration energetics

• Defect mobility  $(\mu)$  is a thermally-activated process:

$$\mu = \mu_0 \exp\!\left(-rac{\mathrm{E_a}}{\mathrm{RT}}
ight)$$

• interstitial sites are higher energy than vacancies, so will be more mobile.



## Variation with temperature

As  $\sigma = nq\mu$  and  $\mu$  is thermally-activated,

$$egin{aligned} \sigma &= n q \mu_0 \exp igg( -rac{ ext{E}_{ ext{a}}}{ ext{RT}} igg) \ &= A \exp igg( -rac{ ext{E}_{ ext{a}}}{ ext{RT}} igg) \end{aligned}$$

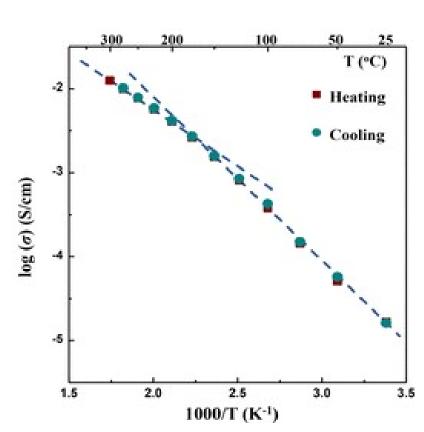
## Variation with temperature

As  $\sigma = nq\mu$  and  $\mu$  is thermally-activated,

$$egin{aligned} \sigma &= n q \mu_0 \exp igg( -rac{ ext{E}_{ ext{a}}}{ ext{RT}} igg) \ &= A \exp igg( -rac{ ext{E}_{ ext{a}}}{ ext{RT}} igg) \end{aligned}$$

Plotting  $\ln \sigma$  vs.  $\frac{1}{T}$  should give a straight line

- more commonly we plot  $\log_{10} \sigma$  vs.  $\frac{1000}{T}$  for high temperature measurements
- gradient is  $\frac{-E_a}{R}$  (or  $\frac{-E_a}{2303R}$  using base 10).



### Lecture recap

- Defects can give rise to ionic conduction
  - Occurs by three main mechanisms:
    - Vacancy hopping
    - Interstitial hopping
    - interstitialcy (knock-on) cooperation
- Ionic conductivity is thermally-activated
  - shows Arrhenius-like behaviour
- Different defects have different conduction energetics
  - Pathways can sometimes be determined experimentally

## Feedback

Return to course contents 21