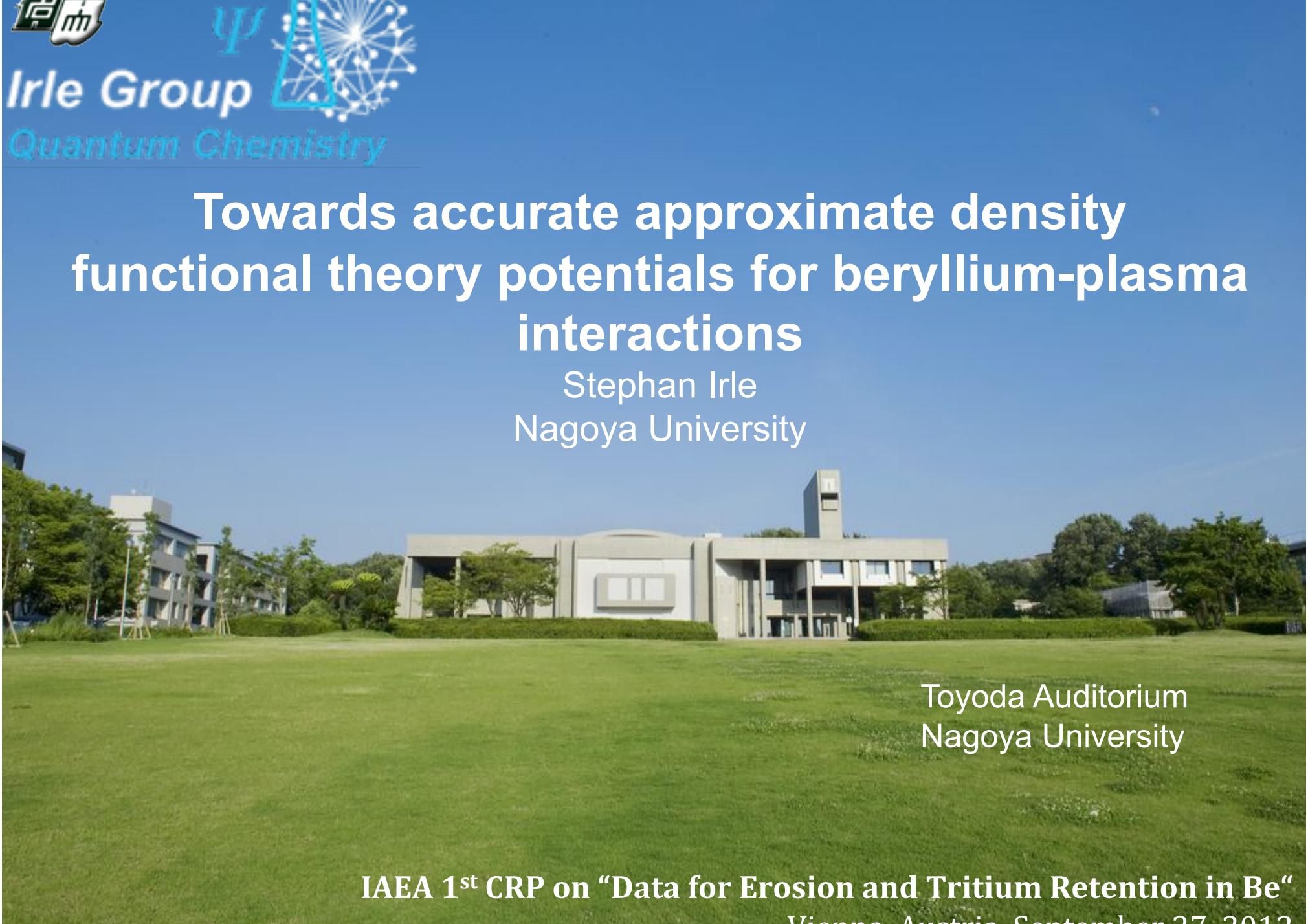




Towards accurate approximate density functional theory potentials for beryllium-plasma interactions

Stephan Irle
Nagoya University

A wide-angle photograph of a modern architectural complex. In the foreground is a large, well-maintained green lawn. Behind it is a long, low-profile building with a light-colored, possibly concrete or stone, facade. A prominent feature is a large, rectangular glass window or entrance area. Further back is a taller, more vertical building with a similar light-colored facade and a visible entrance. The entire complex is set against a clear, bright blue sky. The overall impression is one of a clean, academic, or scientific institution.

Toyoda Auditorium
Nagoya University

IAEA 1st CRP on “Data for Erosion and Tritium Retention in Be”
Vienna, Austria, September 27, 2012

The Group

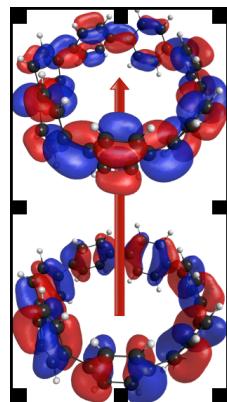
August 9, 2012



Quantum Chemistry of Complex Systems in Nagoya

Collaborator:

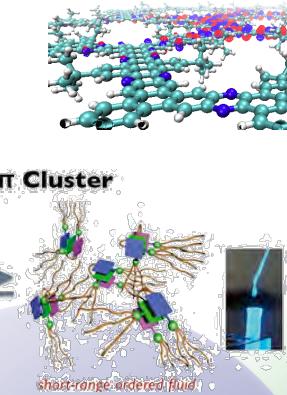
Yamaguchi (expt)
Itami (expt)
Others (expt)



Excited states of
large organic
molecules and
ensembles

$$i\hbar \frac{dP(t)}{dt} = [P(t), F(t)]$$

Collaborator:
Witek (theory)
Jakowski, Niehaus (theory)



QM/MM of
nanostructure self-
assembly and erosion
Structure, properties,
reactivity

Approximate DFT
& *ab initio* methods

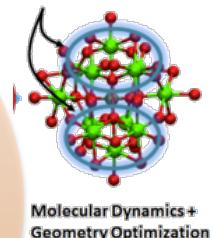
Method
Development:
DFTB parameter
LvNMD, GA
algorithms

Simulation of
ionic liquids:
dynamics and
catalytic activity

Collaborator:
Ouchi (expt)
Parasuk/Hannongbua (theory)

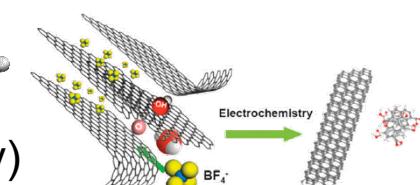
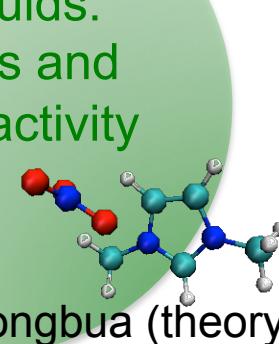
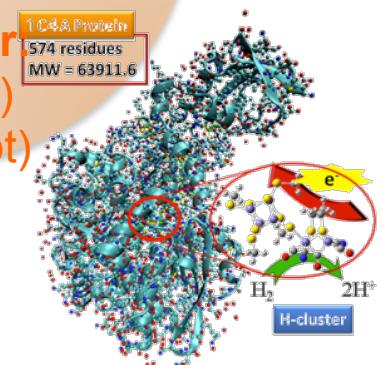


Collaborator:
Shinohara (expt)
Kusunoki (expt)
Jiang (IMS, expt)
Many others (expt)



Transition metal
cluster chemistry:
Materials and
quantum proteins

Collaborator:
Awaga (expt)
Tatsumi (expt)
Endo (expt)



Overview

Motivation

DFTB Method

DFTB/MD Applications

DFTB Parameterization

Summary

Overview

Motivation

DFTB Method

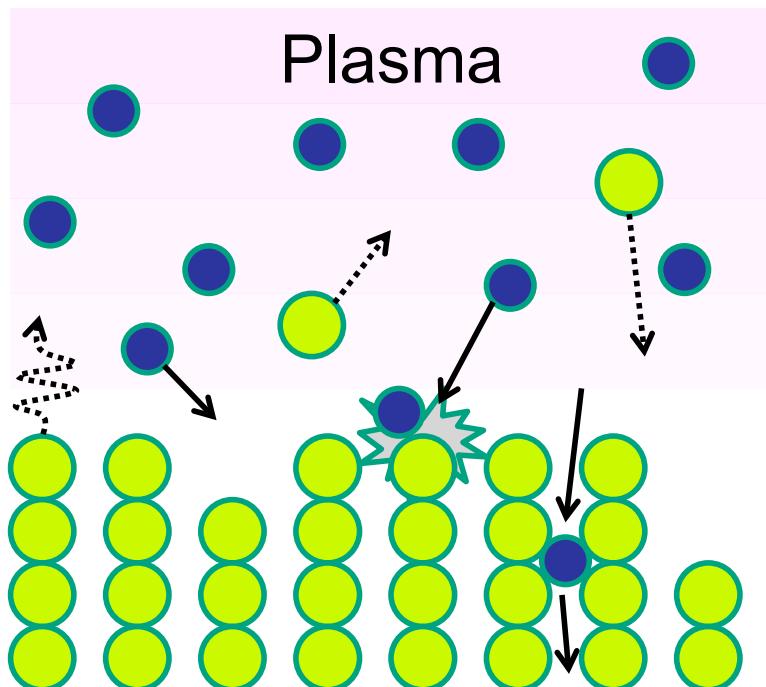
DFTB/MD Applications

DFTB Parameterization

Summary

- Little is known about the plasma wall processes on Be surface
- Performing accurate molecular dynamics (MD) simulations is important to extend our knowledge at the atomistic level

Temperature ~ 600 °C



(pure/mixed) Be surfaces

Yellow circle = Be

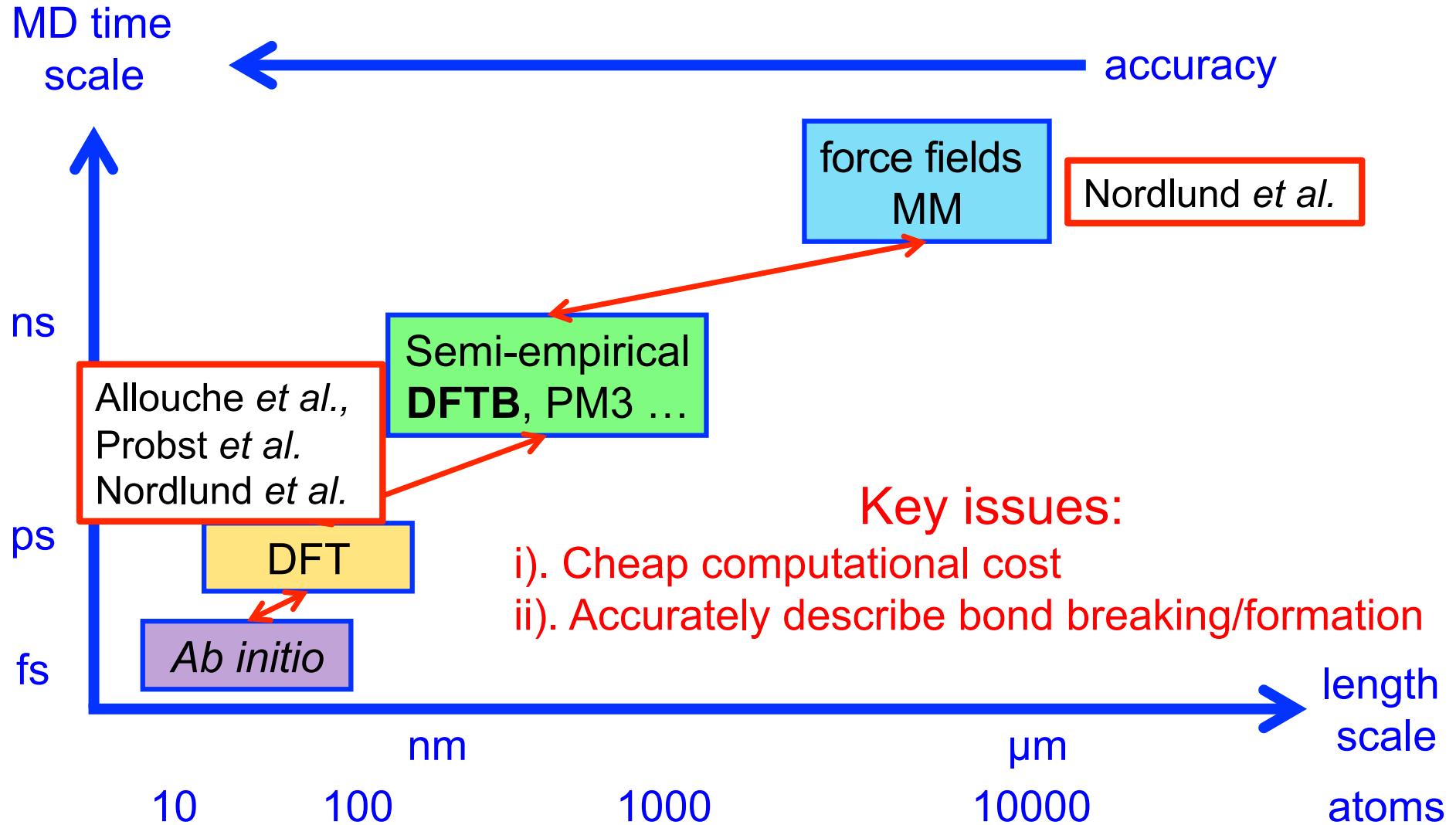
Blue circle = H/D/T, He, H₂O, CH₄, N₂, W,

Getting insights into plasma wall interactions;

- Incident energy dependence
(adsorption, reflection, penetration)
- Temperature dependence
- Conditions of surface structures
(how erosion, retention processes proceed)
- Comparison with experimental data and previous theoretical works (classical MD simulations* and static DFT calculations**)

*Bjorkas, C. et al., *New. J. Phys.*, **11**, 123017, (2009).; Ueda, S. et al., *J. Nucl. Mater.*, **258-263**, 713, (1998).; Ueda, S. et al., *J. Nucl. Mater.*, **283-287**, 1100, (2000).

Allouche, A. et al., *J. Phys. Chem. C*, **114, 3588, (2010).



➤ DFTB (density-functional tight-binding) is a well established **approximate DFT method** but there are *no available Be-X parameters*. Therefore, we started to develop Be-Be and Be-H parameters at first.

Overview

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Summary

Alternative to DFT: Approximate DFT

Density-Functional Tight-Binding: Method using atomic parameters from DFT (PBE, GGA-type), diatomic repulsive potentials from B3LYP

- Seifert, Eschrig (1980-86): minimum STO-LCAO; **2-center approximation**
- Porezag, Frauenheim, *et al.* (1995): efficient parameterization scheme: **NCC-DFTB**
- Elstner *et al.* (1998): charge self-consistency: **SCC-DFTB**
- Köhler *et al.* (2001): spin-polarized DFTB: **SDFTB**



Gotthard
Seifert



Thomas
Frauenheim



Helmut
Eschrig

$$E^{(NCC-)DFTB} = \sum_i^{valence\ orbitals} n_i \varepsilon_i + \frac{1}{2} \sum_{A \neq B}^{atoms} E_{AB}^{rep}$$

$$E^{(SCC-)DFTB} = E^{(NCC-)DFTB} + \frac{1}{2} \sum_{A \neq B}^{atoms} \gamma_{AB} \Delta q_A \Delta q_B$$

Linear response:
TD-DFTB



Marcus Elstner



Thomas
Niehaus

$$E^{S(pin-polarized)DFTB} = E^{(SCC-)DFTB} + \frac{1}{2} \sum_A \sum_{l \in A} \sum_{l' \in A} p_{Al} p_{Al'} W_{All'}$$



Christof Köhler

Taken from Oliveira, Seifert, Heine, Duarte, *J. Braz. Chem. Soc.*
20, 1193-1205 (2009)



Thomas
Heine



Helio
Duarte

J. Braz. Chem. Soc., Vol. 20, No. 7, 1193-1205, 2009.
Printed in Brazil - ©2009 Sociedade Brasileira de Química
0103 - 5053 \$6.00+0.00

Review

Density-Functional Based Tight-Binding: an Approximate DFT Method

Augusto F. Oliveira,^{*,a,b} Gotthard Seifert,^b Thomas Heine^c and Hélio A. Duarte^{*,a}

^aDepartamento de Química, Instituto de Ciências Exatas, Universidade Federal de Minas Gerais,
Av. Antonio Carlos, 6627, 31270-901 Belo Horizonte-MG, Brazil

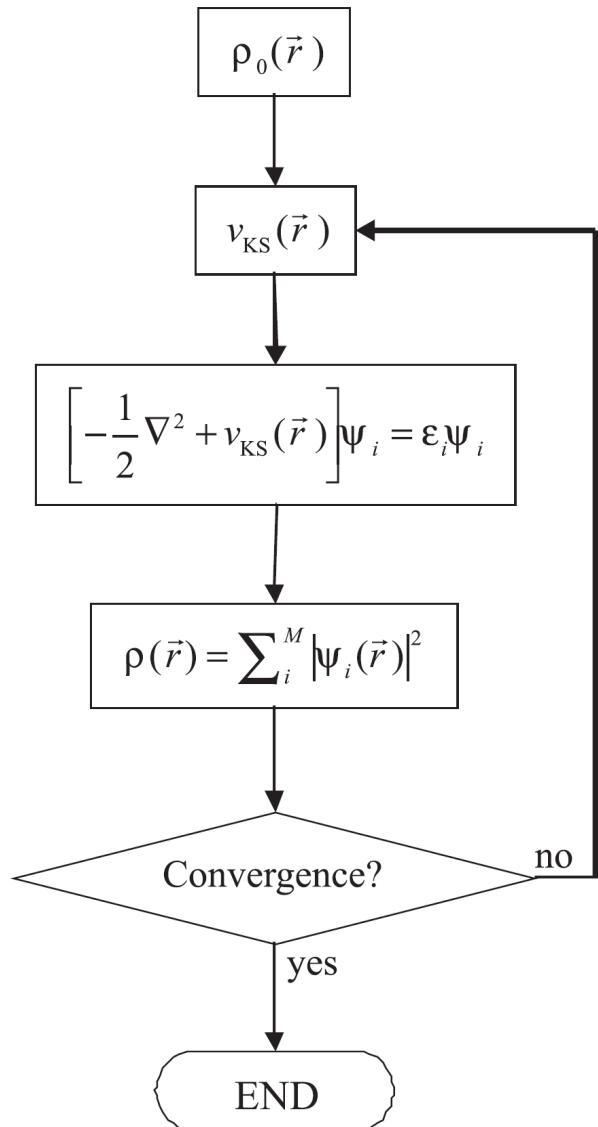
^bPhysikalische Chemie, Technische Universität Dresden, Mommsenstr, 13, D-01062 Dresden, Germany

^cSchool of Engineering and Sciences, Jacobs University, P.O. Box 750 561, 28725 Bremen, Germany

...open access



Density Functional Theory (DFT)



Walter Kohn/John A. Pople



1998

at convergence:

$$\begin{aligned}
 E[\rho] &= \sum_{i=1}^M n_i \langle \psi_i | -\frac{1}{2} \nabla^2 + v_{ext}(\vec{r}) + \int \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3 r' | \psi_i \rangle \\
 &\quad + E_{xc}[\rho] - \frac{1}{2} \int \int \frac{\rho(\vec{r}) \rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3 r d^3 r' + \frac{1}{2} \sum_{\substack{\alpha, \beta=1 \\ \alpha \neq \beta}}^N \frac{Z_\alpha Z_\beta}{|\vec{R}_\alpha - \vec{R}_\beta|}
 \end{aligned}$$

$$= \sum_{i=1}^M n_i \epsilon_i + E_{rep}$$

Various criteria for convergence possible:

- Electron density
- Potential
- Orbitals
- Energy
- Combinations of above quantities

Foulkes + Haydock Ansatz

Phys. Rev. B, 39, 12520 (1989)

$$\rho(\vec{r}) = \rho_0(\vec{r}) + \delta\rho(\vec{r})$$

$$\begin{aligned}
 E[\rho_0 + \delta\rho] = & \sum_i^M n_i \left\langle \Psi_i \left| -\frac{1}{2} \nabla^2 + v_{\text{ext}}(\vec{r}) + \int \frac{\rho'_0}{|\vec{r} - \vec{r}'|} d\vec{r}' + v_{\text{xc}}[\rho_0] \right| \Psi_i \right\rangle \\
 & - \frac{1}{2} \iint \frac{\rho'_0(\rho_0 + \delta\rho)}{|\vec{r} - \vec{r}'|} d\vec{r} d\vec{r}' - \int v_{\text{xc}}[\rho_0](\rho_0 + \delta\rho) d\vec{r} \\
 & + \frac{1}{2} \iint \frac{\delta\rho'(\rho_0 + \delta\rho)}{|\vec{r} - \vec{r}'|} d\vec{r} d\vec{r}' + \boxed{E_{\text{xc}}[\rho_0 + \delta\rho]} + E_{\text{nn}}
 \end{aligned}$$

Second-Order Taylor Expansion

Elstner et al. *Phys. Rev. B*: 58, 7260 (1998)

$$E_{\text{xc}}[\rho_0 + \delta\rho] = E_{\text{xc}}[\rho_0] + \int \frac{\delta E_{\text{xc}}}{\delta\rho} \Bigg|_{\rho_0} \delta\rho d\vec{r} + \frac{1}{2} \iint \frac{\delta^2 E_{\text{xc}}}{\delta\rho \delta\rho'} \Bigg|_{\rho_0} \delta\rho \delta\rho' d\vec{r} d\vec{r}' \quad (17)$$

Note that $\delta E_{\text{xc}}/\delta\rho)_{\rho_0} = v_{\text{xc}}[\rho_0]$

DFT 2nd Order Energy Expression

Elstner et al. Phys. Rev. B: 58, 7260 (1998)

$$E = \sum_i^M n_i \left\langle \Psi_i \left| -\frac{1}{2} \nabla^2 + v_{\text{ext}}(\vec{r}) + \int \frac{\rho_0'}{|\vec{r} - \vec{r}'|} d\vec{r}' + v_{\text{xc}}[\rho_0] \right| \Psi_i \right\rangle$$

$$- \frac{1}{2} \iint \frac{\rho_0' \rho_0}{|\vec{r} - \vec{r}'|} d\vec{r} d\vec{r}' + E_{\text{xc}}[\rho_0] - \int v_{\text{xc}}[\rho_0] \rho_0 d\vec{r} + E_{\text{nn}}$$

$$+ \frac{1}{2} \iint \left(\frac{\delta \rho \delta \rho'}{|\vec{r} - \vec{r}'|} + \frac{\delta^2 E_{\text{xc}}}{\delta \rho \delta \rho'} \Big|_{\rho_0} \right) d\vec{r} d\vec{r}'$$

$$E[\rho_0, \delta \rho] = \sum_{i=1}^M n_i \langle \psi_i | \hat{H}_0 | \psi_i \rangle + E_{\text{rep}} + E_{\text{2nd}}$$

Approximations

- Get the **best possible ρ_0** – atomic densities, new ideas possible, e.g. fragment densities etc....
- Reduce **number of basis functions** as much as possible – minimum valence basis set (atomic orbitals, optimized local orbitals)
- Ignore (DFTB) or approximate (e.g. as in SCC-DFTB)
 2^{nd} order terms, (add 3^{rd} order terms, DFTB3)
- Introduce **further approximations in the Hamiltonian**

Approximations

Traditional DFTB concept: Hamiltonian matrix elements are approximated to **two-center terms**. The same types of approximations are done to E_{rep} .

$$H_{\mu\nu}^0 = \begin{cases} \varepsilon_{\mu}^{\text{neutral free atom}} & \text{if } \mu = \nu \\ \langle \varphi_{\mu}^{\alpha} | \hat{T} + V_0^{\alpha} + V_0^{\beta} | \varphi_{\nu}^{\beta} \rangle & \text{if } \alpha \neq \beta \\ 0 & \text{otherwise.} \end{cases}$$

From Elstner et al., PRB 1998

$$V_{\text{eff}}[\rho_0] \approx V_{\text{eff}}[\rho_A + \rho_B]$$

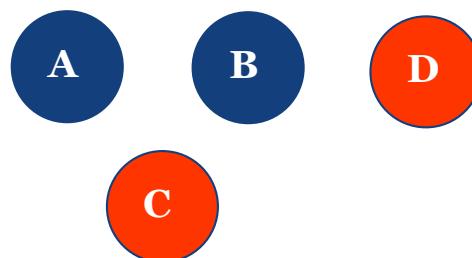
(Density superposition)

$$V_{\text{eff}}[\rho_0] \approx V_{\text{eff}}[\rho_A] + V_{\text{eff}}[\rho_B]$$

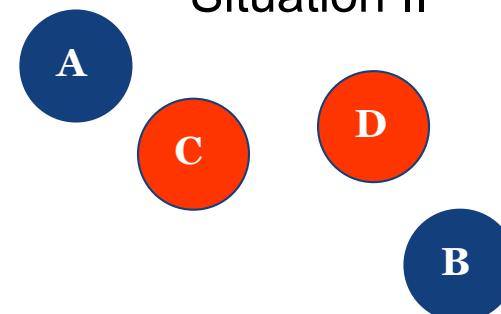
(Potential superposition)

Both approximations are justified by the screening argument: Far away, neutral atoms have no Coulomb contribution.

Situation I



Situation II



Approximations

Approximations in SCC-DFTB (2nd order expansion only) potential:

$$\delta\rho = \sum_{\alpha}^N \delta\rho_{\alpha} \quad \delta\rho_{\alpha} \approx \Delta q_{\alpha} F_{00}^{\alpha} Y_{00}$$

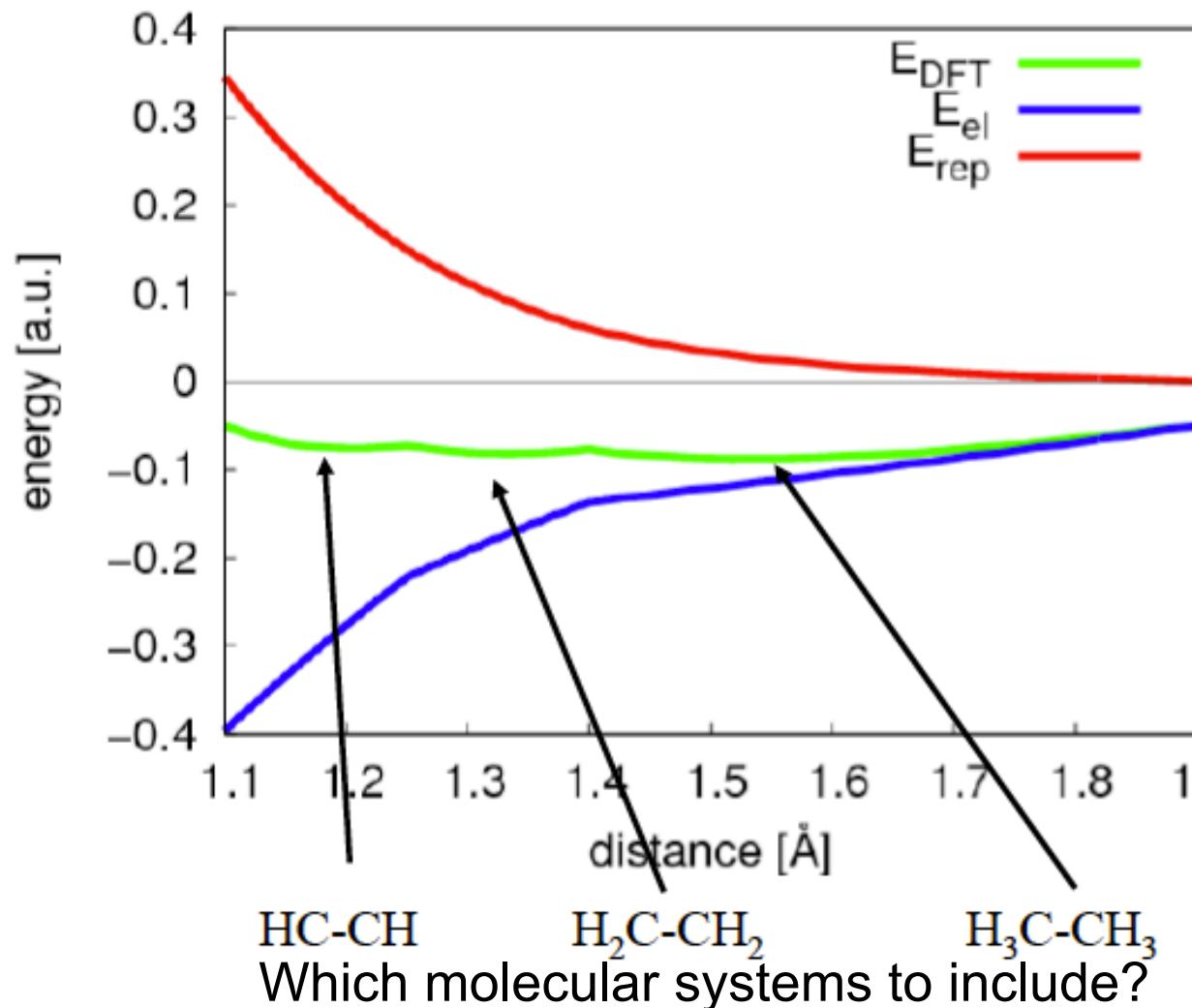
$$E_{2nd} \approx \frac{1}{2} \sum_{\alpha, \beta}^N \Delta q_{\alpha} \Delta q_{\beta} \underbrace{\iint \left(\frac{1}{|\vec{r} - \vec{r}'|} + \frac{\delta^2 E_{xc}}{\delta \rho \delta \rho'} \Big|_{\rho_0} \right) F_{00}^{\alpha} F_{00}^{\beta} Y_{00}^2 d\vec{r} d\vec{r}'}_{\gamma_{\alpha\beta}}$$

$$U_{Ai} = \frac{\partial^2 E_{DFT}^{total}}{\partial n_i^2} = \frac{\partial \epsilon_i}{\partial n_i}$$

Note that directional changes of the density are not covered with this approximation.

DFTB repulsive potential E_{rep}

$$E[\rho] = \sum_i^{\text{occ}} \epsilon_i^H + \sum_{\alpha\beta} \mathbf{U}_{\alpha\beta} \quad \rightarrow \quad \mathbf{U}_{\alpha\beta} = E^{\text{DFT}}[\rho](R_{\alpha\beta}) - \sum_i^{\text{occ}} \epsilon_i^H(R_{\alpha\beta})$$



Development of (semi-) automatic fitting:

- Knaup, J. et al., *JPCA*, **111**, 5637, (2007)
- Gaus, M. et al., *JPCA*, **113**, 11866, (2009)
- Bodrog Z. et al., *JCTC*, **7**, 2654, (2011)

SCC-DFTB

1. All contributions to **Hamiltonian and Overlap matrices** are pre-computed.
At runtime, values have to be interpolated (negligible time for computing matrix elements)
2. **Total energy** is simply computed using KS orbital energies (eigenvalues of the Hamiltonian matrix) and repulsive potential (simple short-range two-body force field)
3. Energy **gradients** are easily calculated (analytical derivative of total energy expression)
4. Aiming at large systems, Hamiltonian and Overlap matrices are very sparse. **Sparsity** can be transferred to energy and energy gradient calculations, as density / energy weighted density matrices are required only where other matrices are non-zero (identical sparsity pattern)
5. Typically **1000 times faster than conventional DFT**

Overview

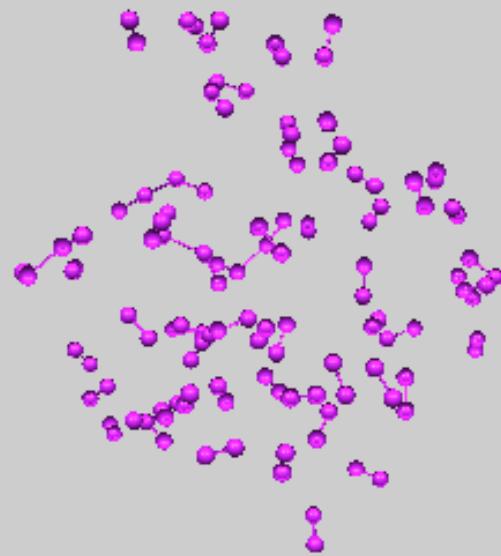
Motivation

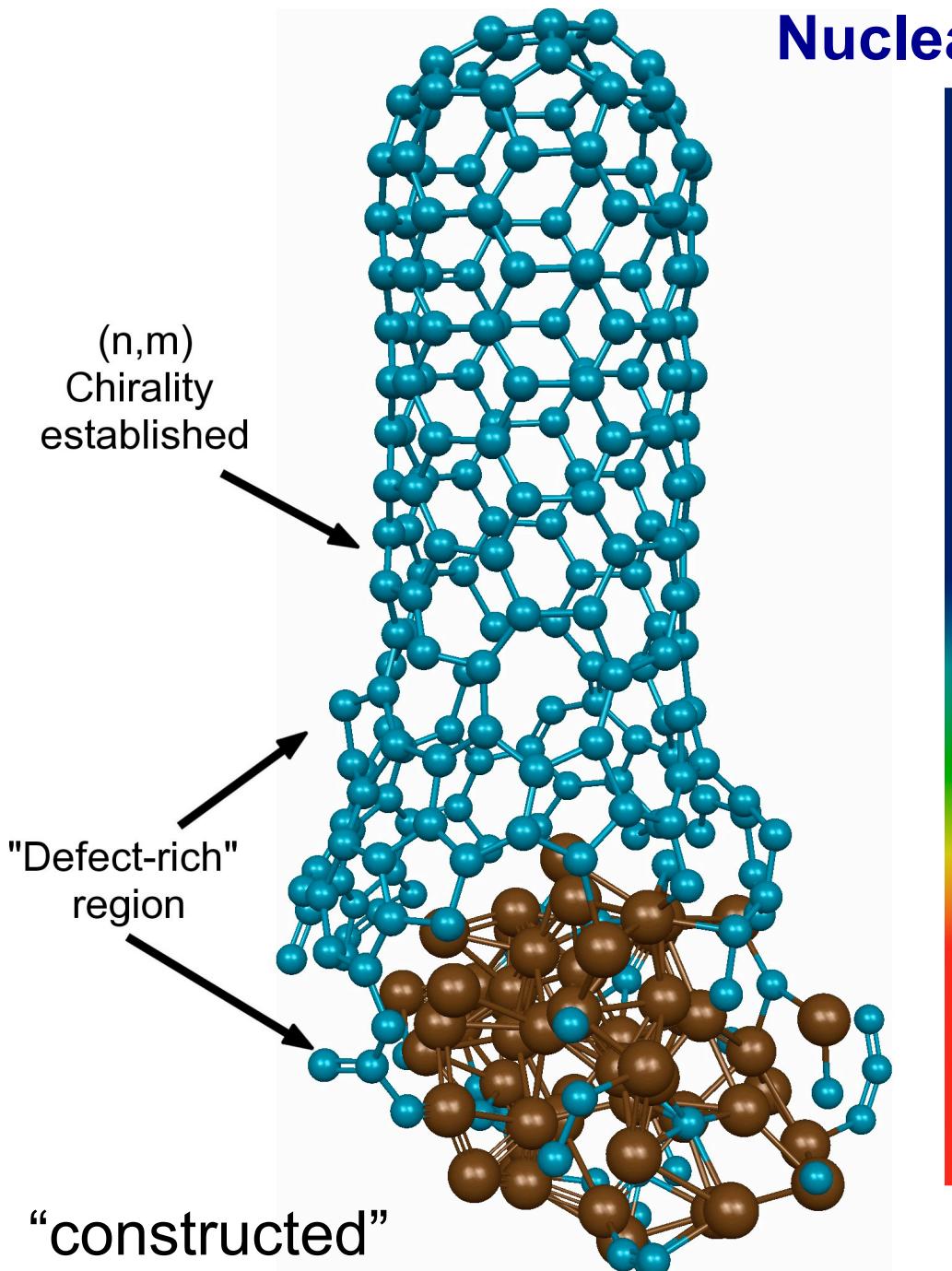
DFTB Method

DFTB/MD Applications

DFTB Parameterization

Summary





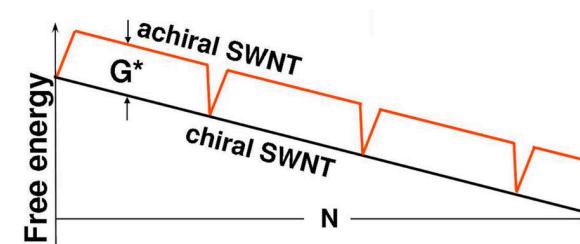
Nucleation and growth hypothesis:

A. J. Page, Y. Ohta, S. Irie, and K. Morokuma,

In sharp contrast to:



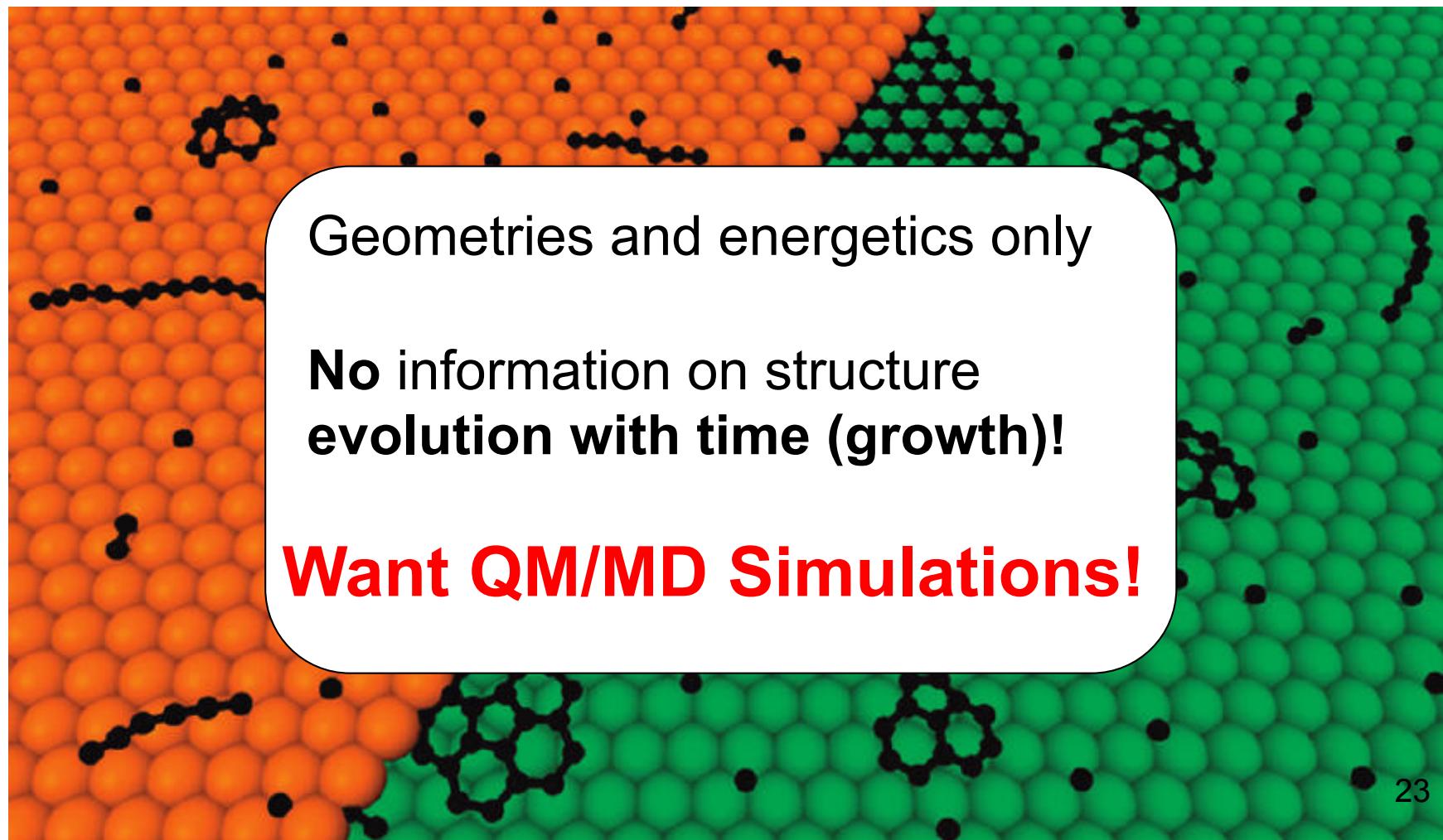
F. Ding, A. Harutyunyan, B. I. Yakobson,
Proc. Natl. Acad. Sci. **106**, 2506 (2009)



How Does Graphene Form on Ni(111)?

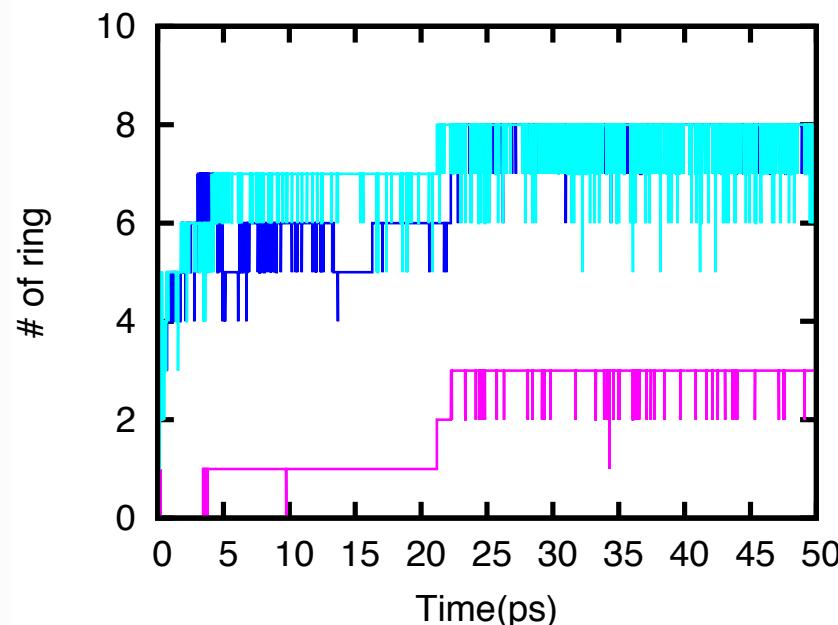
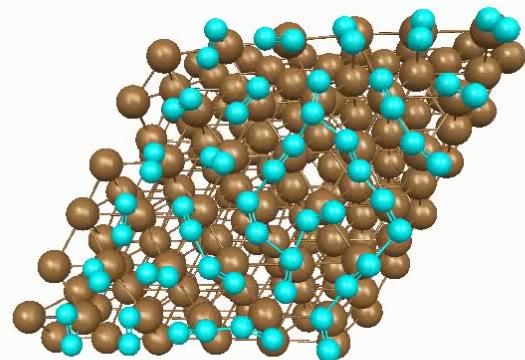
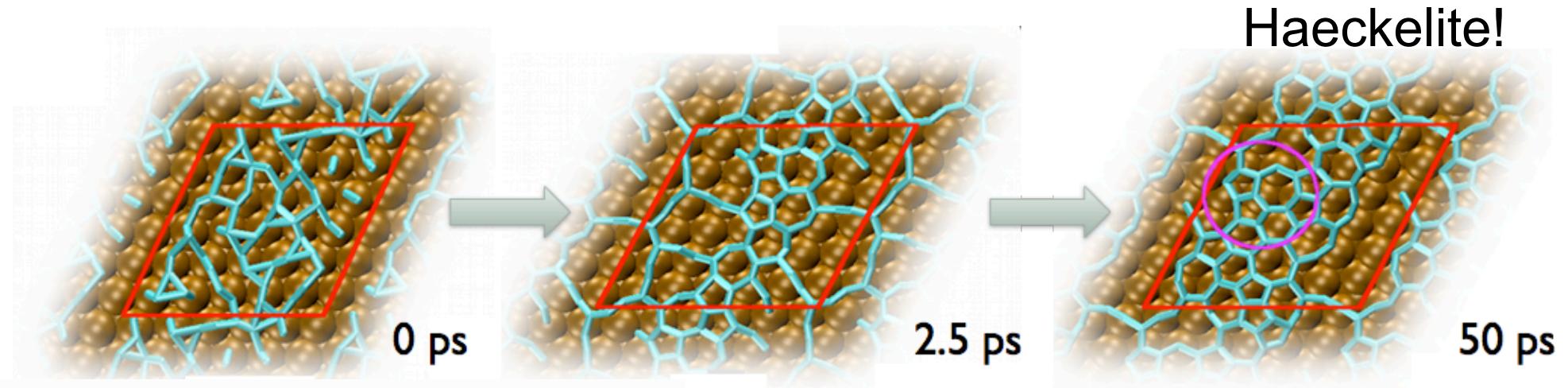
Gao *et al.* J. Am. Chem. Soc. 133, 5009 (2011)

- GGA PW91/UPP-PW (VASP) geometry optimizations
- individual clusters on Ni(111) C1-C24

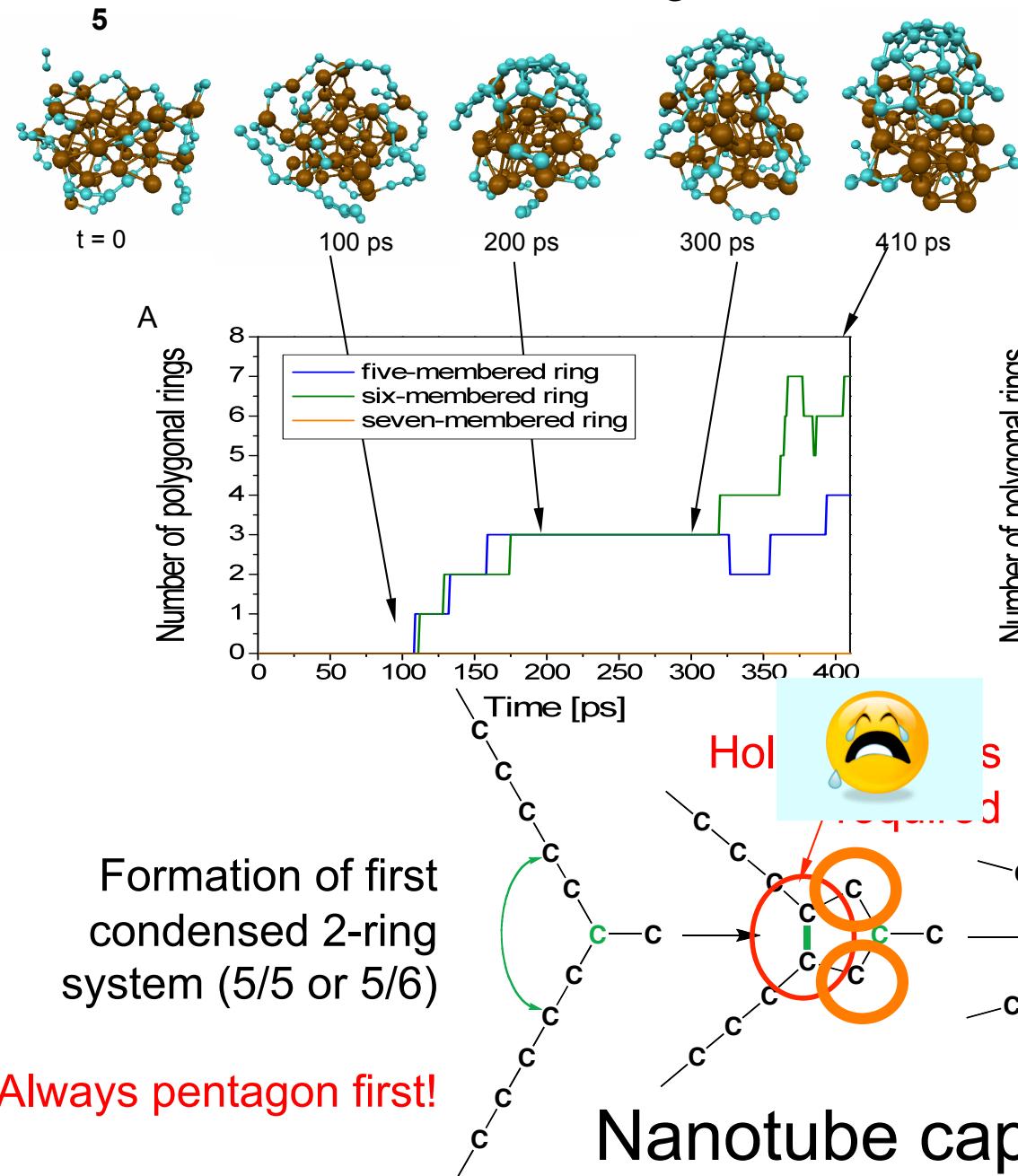


QM/MD of 30 C₂ on Ni(111), 1180 K

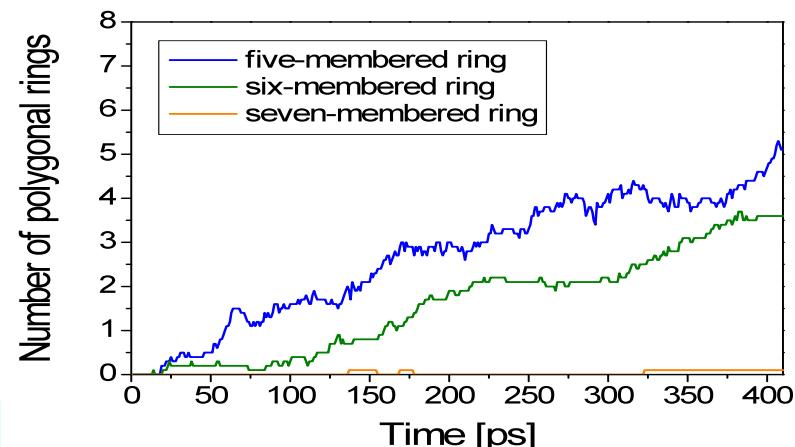
Y. Wang, A. J. Page, Y. Nishimoto, H.-J. Qian, K. Morokuma, SI, JACS (2011)



Y. Ohta, Y. Okamoto, A. J. Page, SI, K. Morokuma, ACS Nano 3, 3413 (2009)

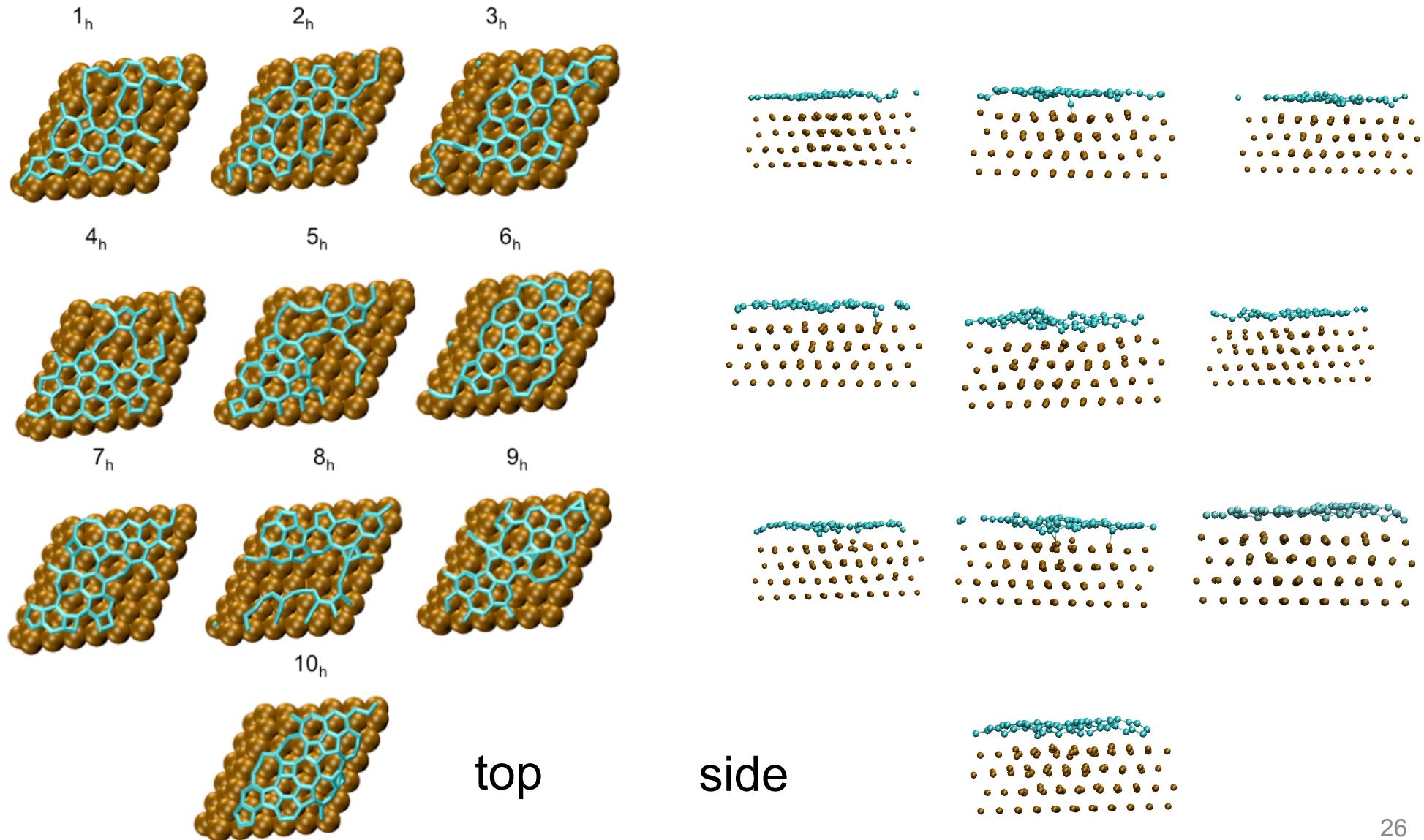


Average 5- and 6-ring counts over 10 annealing trajectories



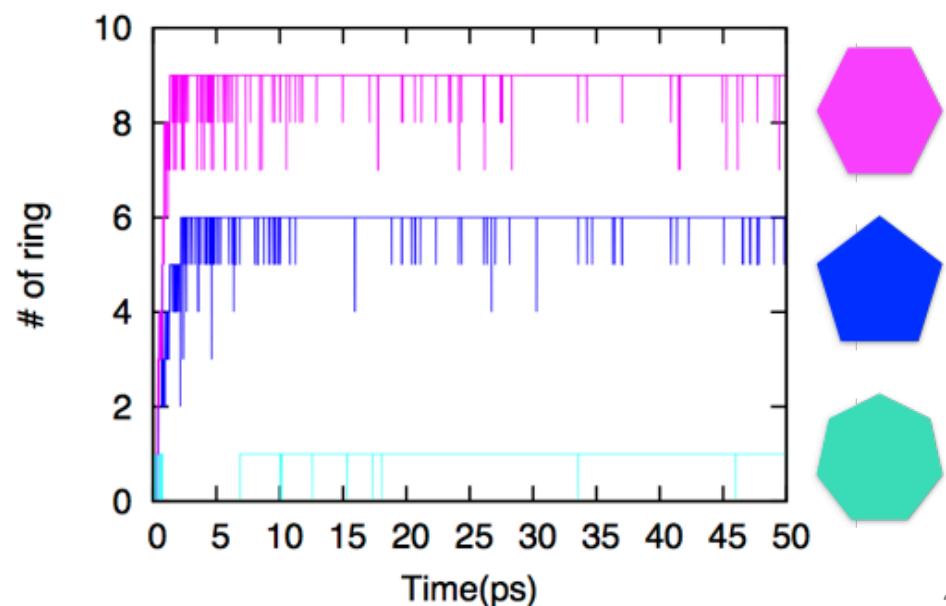
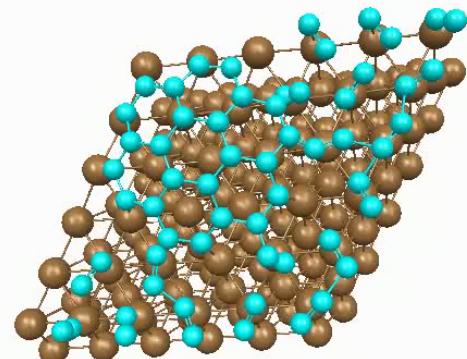
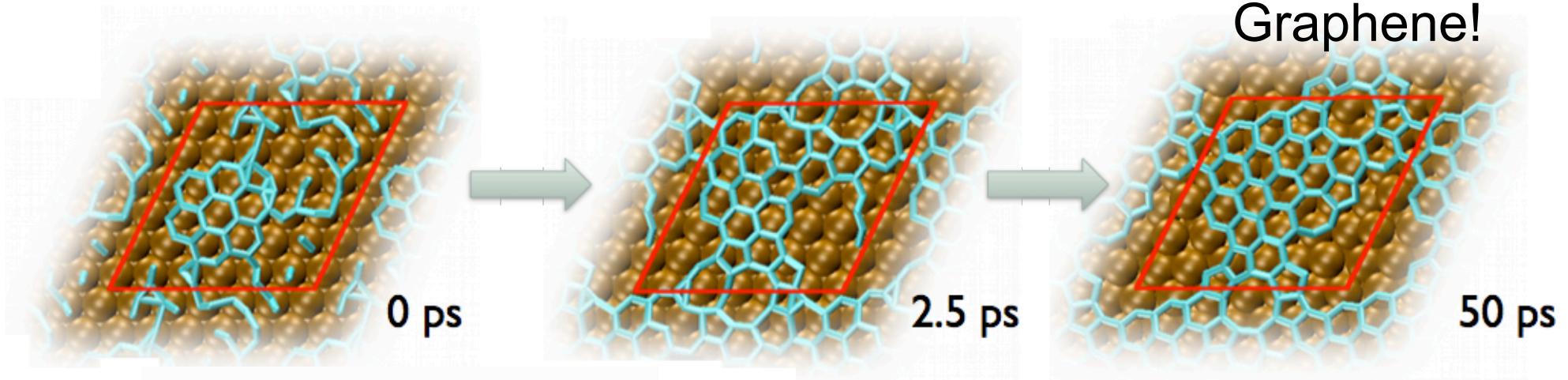
QM/MD of 30 C₂ on Ni(111), 1180 K

Y. Wang, A. J. Page, Y. Nishimoto, H.-J. Qian, K. Morokuma, SI, JACS (2011)



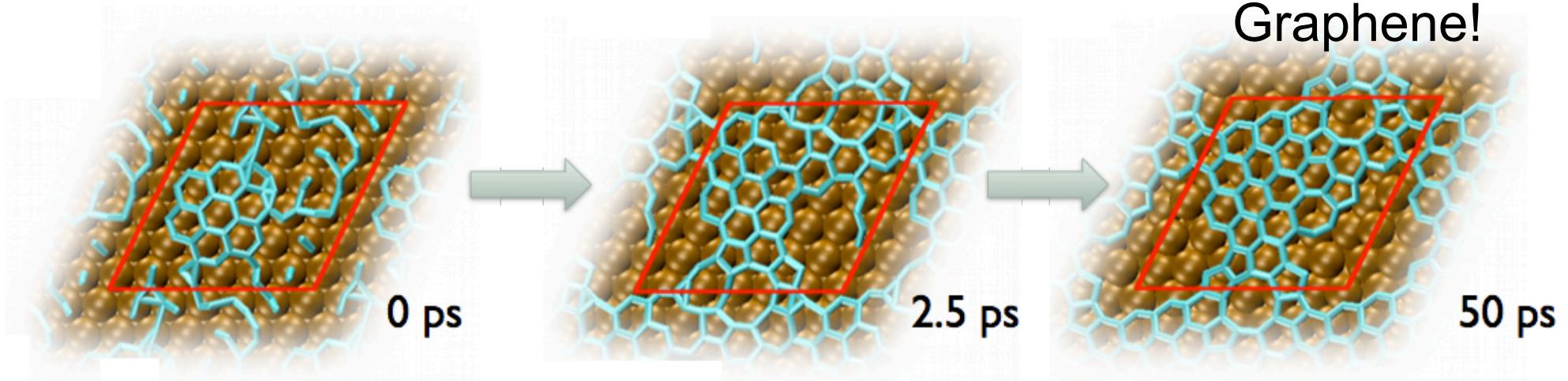
QM/MD of 18 C₂ + C₂₄ on Ni(111), 1180 K

Y. Wang, A. J. Page, Y. Nishimoto, H.-J. Qian, K. Morokuma, SI, JACS (2011)

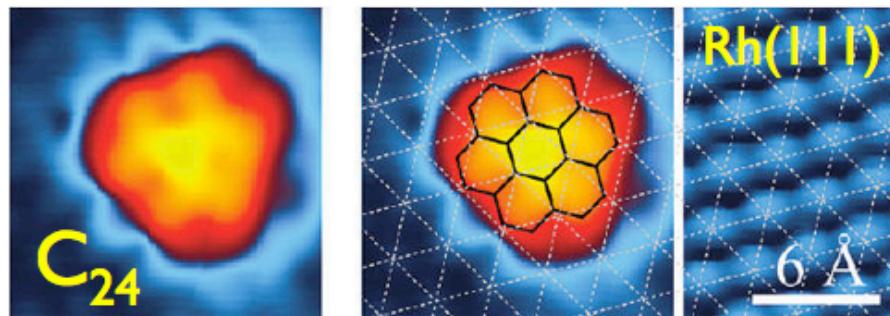


QM/MD of $18 \text{ C}_2 + \text{C}_{24}$ on Ni(111), 1180 K

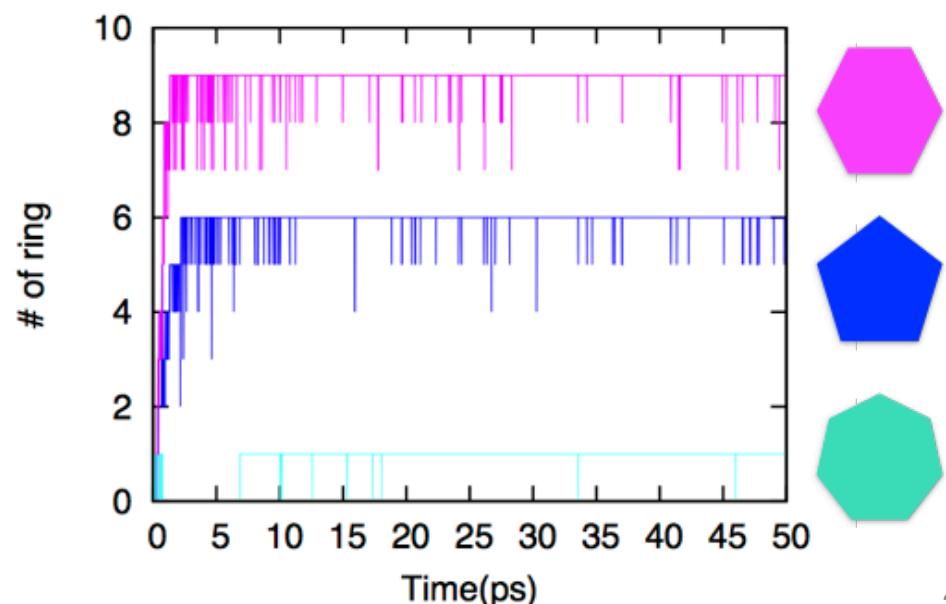
Y. Wang, A. J. Page, Y. Nishimoto, H.-J. Qian, K. Morokuma, SI, JACS (2011)



- Pentagon-first vs. **template effect**.
- Suppression of heptagons and pentagons

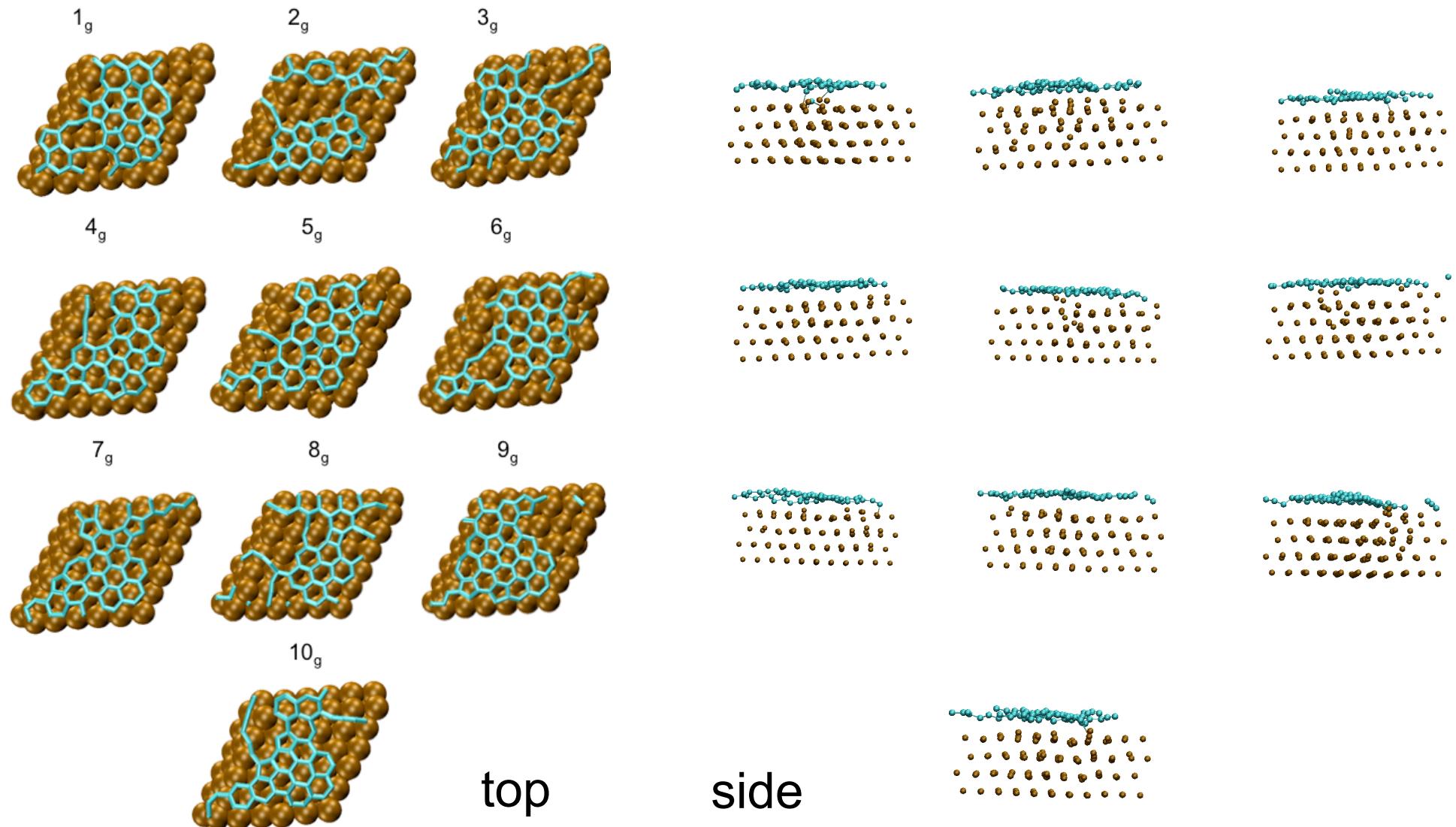


Wang et al., Nano Lett., (2011)

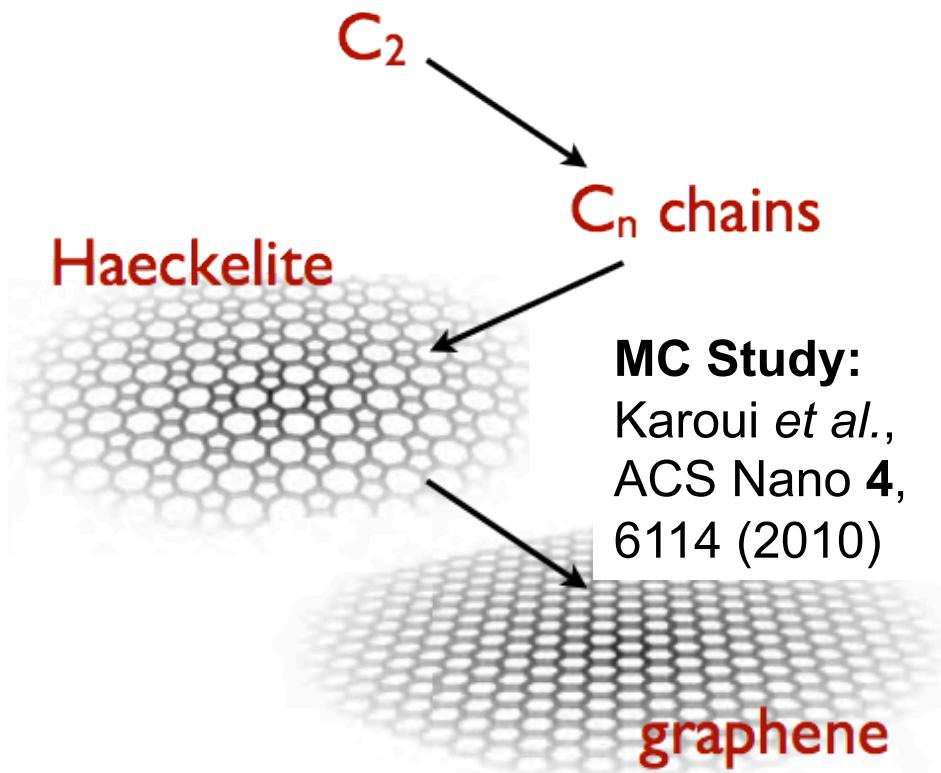


QM/MD of $18 \text{ C}_2 + \text{C}_{24}$ on Ni(111), 1180 K

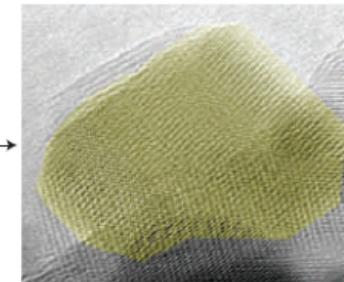
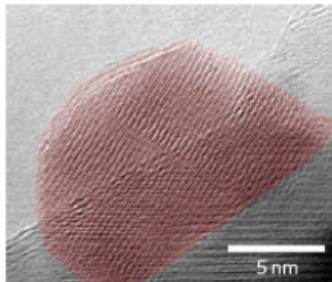
Y. Wang, A. J. Page, Y. Nishimoto, H.-J. Qian, K. Morokuma, SI, JACS (2011)



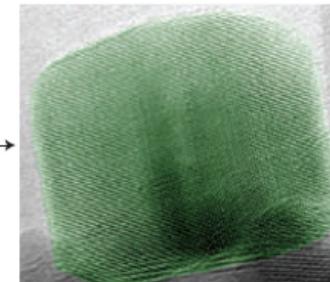
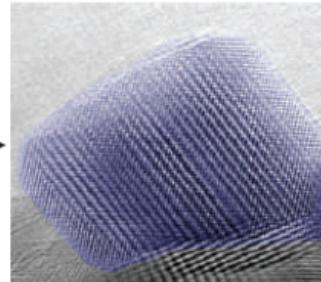
Haeckelite is a Metastable Phase



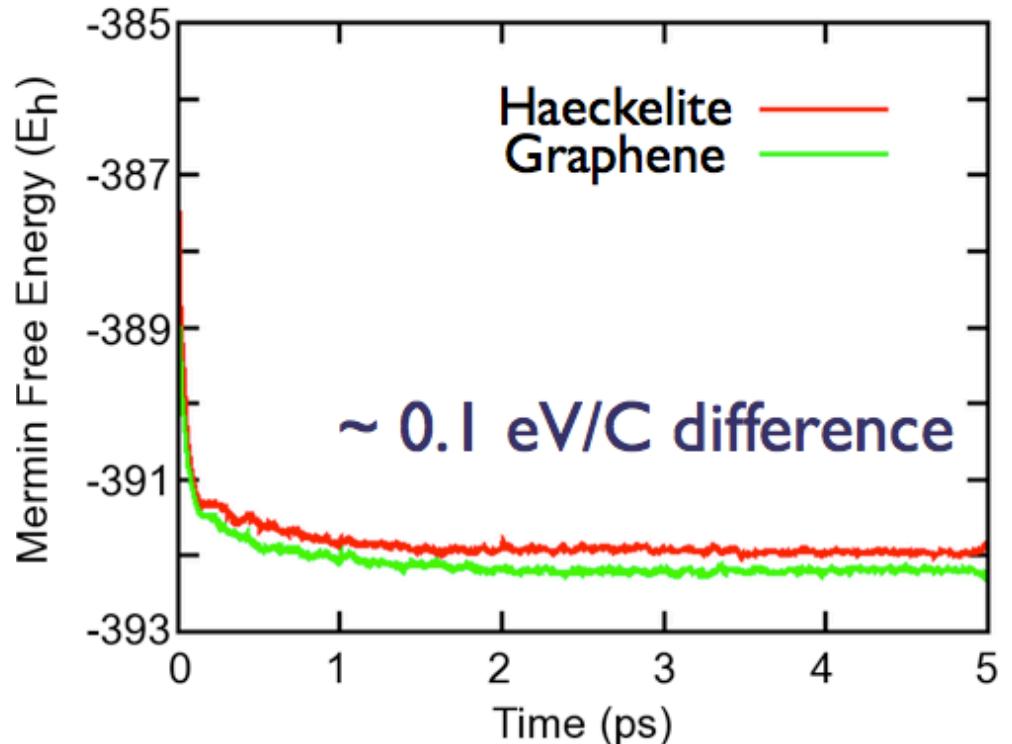
LiFePO₄ nanocrystal, 450 °C



Ostwald's 'rule of stages'



Chung et al., Nature Phys. (2008)



F. W. Ostwald,
Z. Phys. Chem.
22, 289 (1897)³⁰

Other simulations

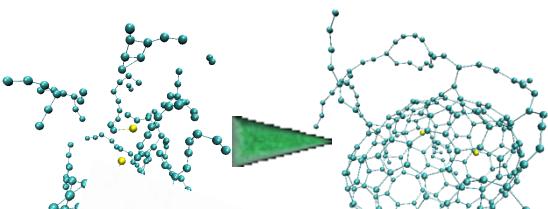
DFTB/MD Applications

Saha et al., ACS Nano 3, 2241 (2009)

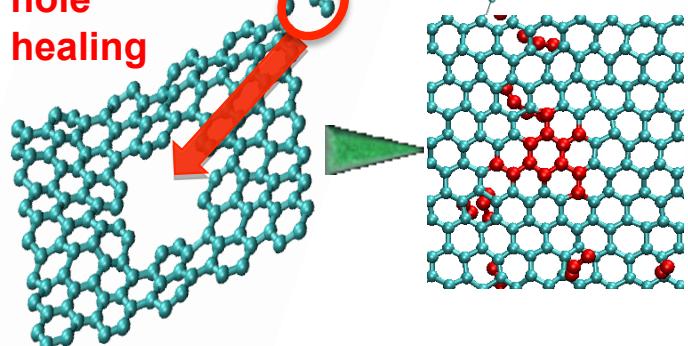
Fullerenes in benzene combustion



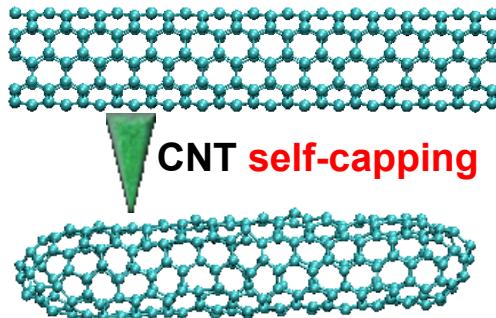
Sc & Ti metallocfullerene formation



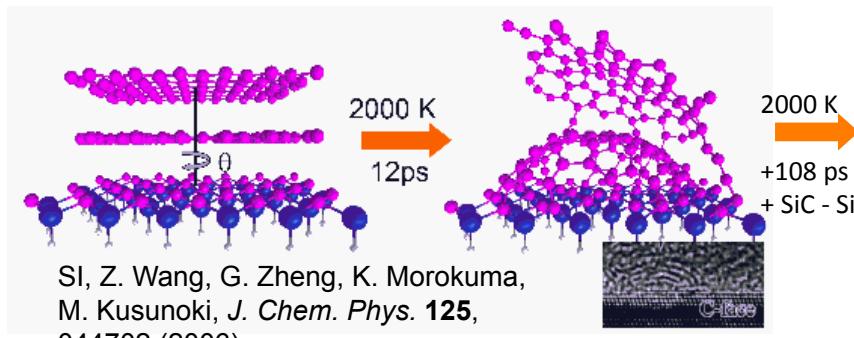
Graphene hole healing



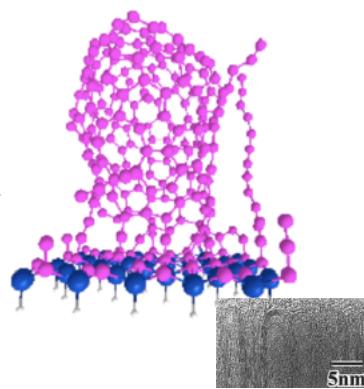
CNT self-capping



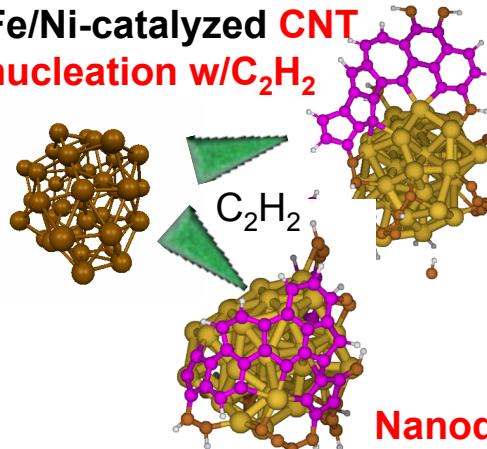
CNT growth on SiC(000-1)



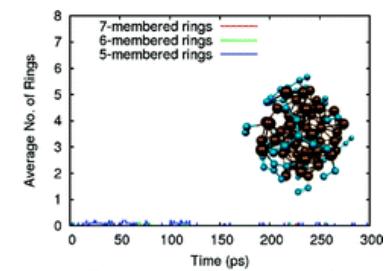
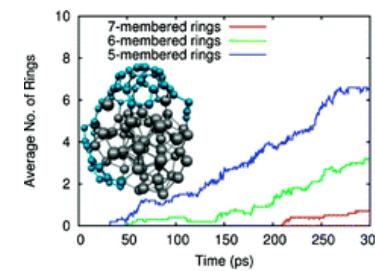
Z. Wang, SI, G. Zheng, M. Kusunoki, K. Morokuma, J. Phys. Chem. C 111, 12960 (2007)



Fe/Ni-catalyzed CNT nucleation w/ C_2H_2



CNT nucleation from metal carbide

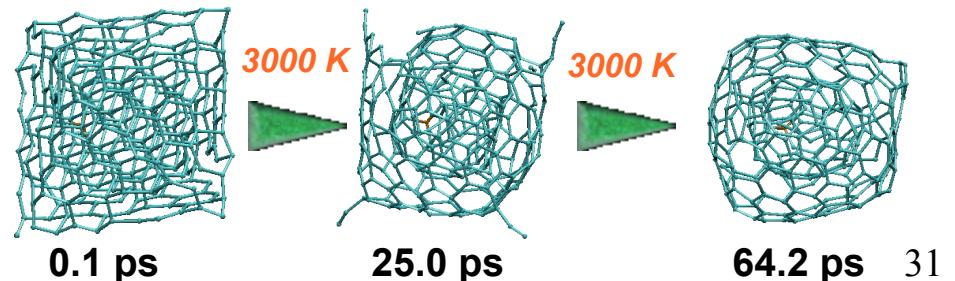


Ni_xC_y : SWNT Nucleation enhanced

Fe_xC_y : SWNT Nucleation impeded

Page et al. J. Am. Chem. Soc. 132, 15699 (2010)

Nanodiamond to Spiroïd to Onion Transformation upon Heating



Methods to propagate molecular systems on a single potential energy surface (PES)

Reactive force fields

Energy and forces on molecules from exp parameters

Nuclei moved:
Newton's laws

(MM)

Born-Oppenheimer MD

Solve electronic Schrödinger Eq.
(convergence)
at nuclear configuration

Nuclei propagated from gradients
(classically)

(BOMD)

Extended Lagrangian MD

Electronic structure *propagated classically:*
not converged

integer occ. numbers:
metals difficult

“Simultaneous”
classical propagation of electrons and nuclei

(CPMD)

Liouville-von Neumann MD

Jakowski, Morokuma,
JCP **130**, 224106 (2009)

Electronic structure:
Quantum dynamics:
mixed states possible

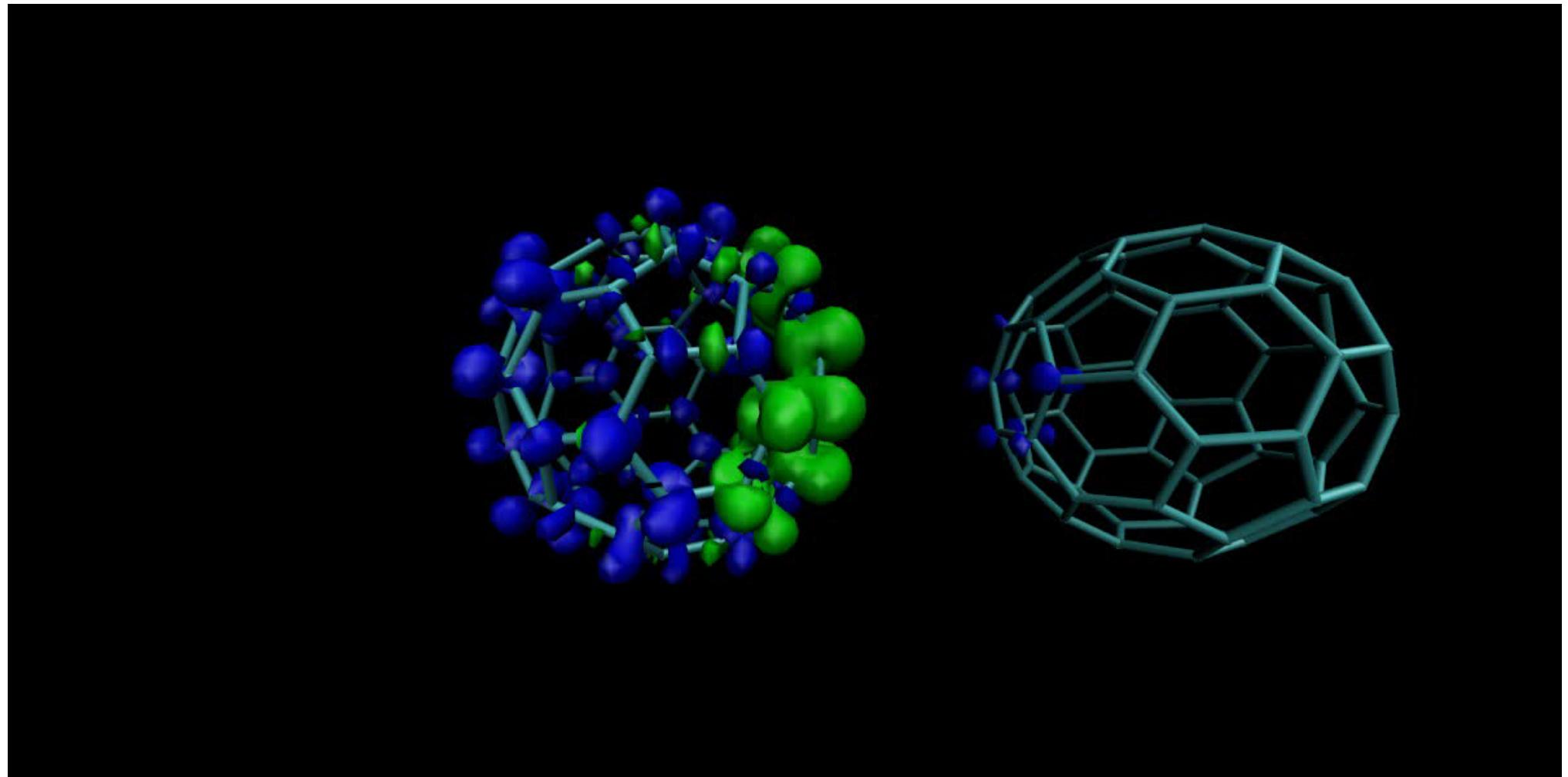
$$i\hbar \frac{\partial}{\partial t} \psi(r; t) = \hat{H} \psi(r; t)$$

Nuclei propagated from gradients
(classically)

(LvNMD)

Collision energy: 1.4 eV C_{60}^+

C_{60}^+



$$\Delta\rho = \rho_{total} - \rho_{C60}$$

Green: $-\Delta\rho$
Blue: $+\Delta\rho$

$$\Delta\rho = \rho_{total} - \rho_{C60}^{2+}$$

Overview

Motivation

DFTB Method

DFTB/MD Applications

DFTB Parameterization

Summary

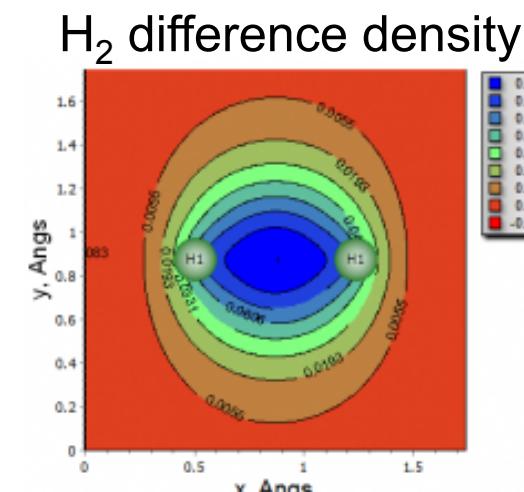
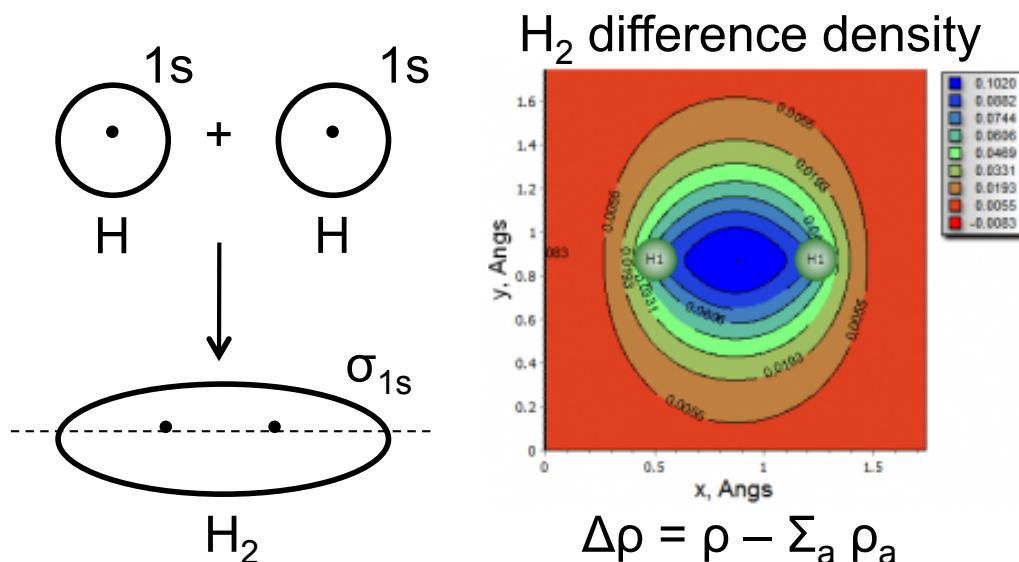
New Confining Potentials

➤ Typically, electron density contracts under covalent bond formation.

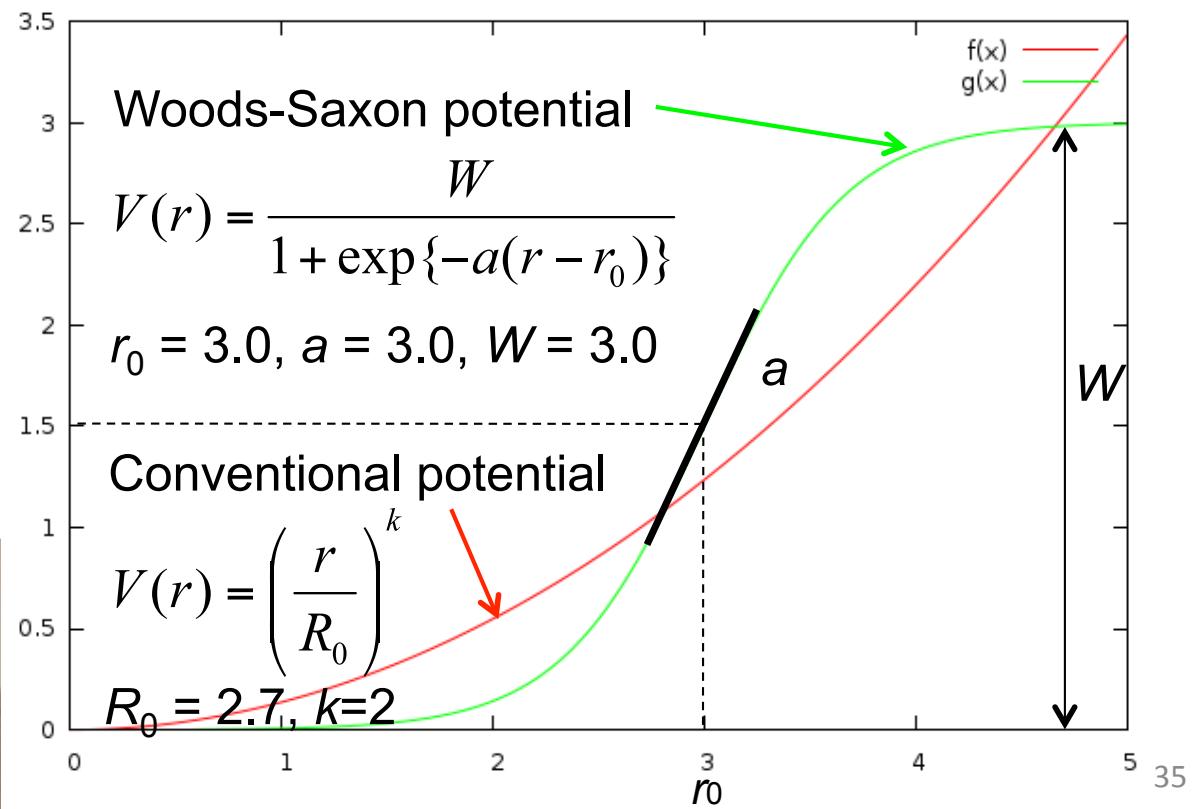
➤ In standard *ab initio* methods, this problem can be remedied by including more basis functions.

➤ DFTB uses minimal valence basis set: the confining potential is adopted to mimic contraction

Henryk Witek



$$\Delta\rho = \rho - \sum_a \rho_a$$



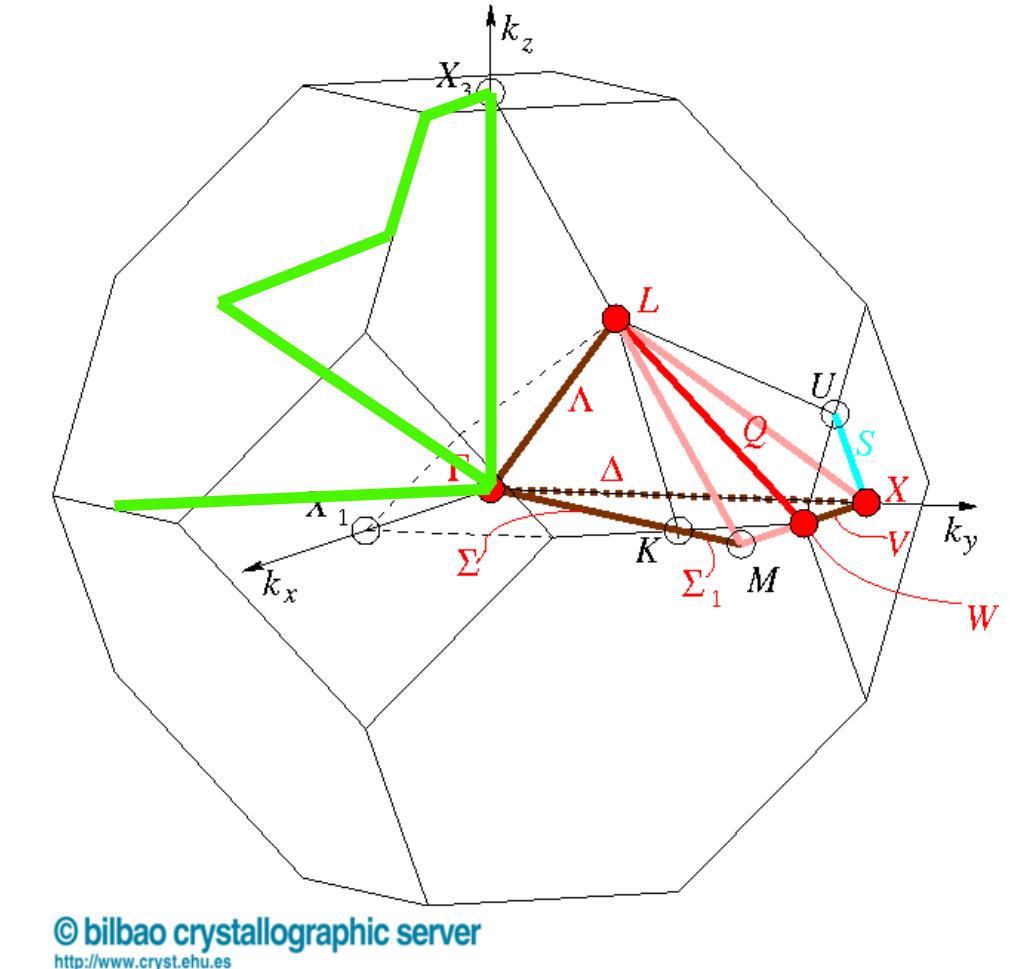
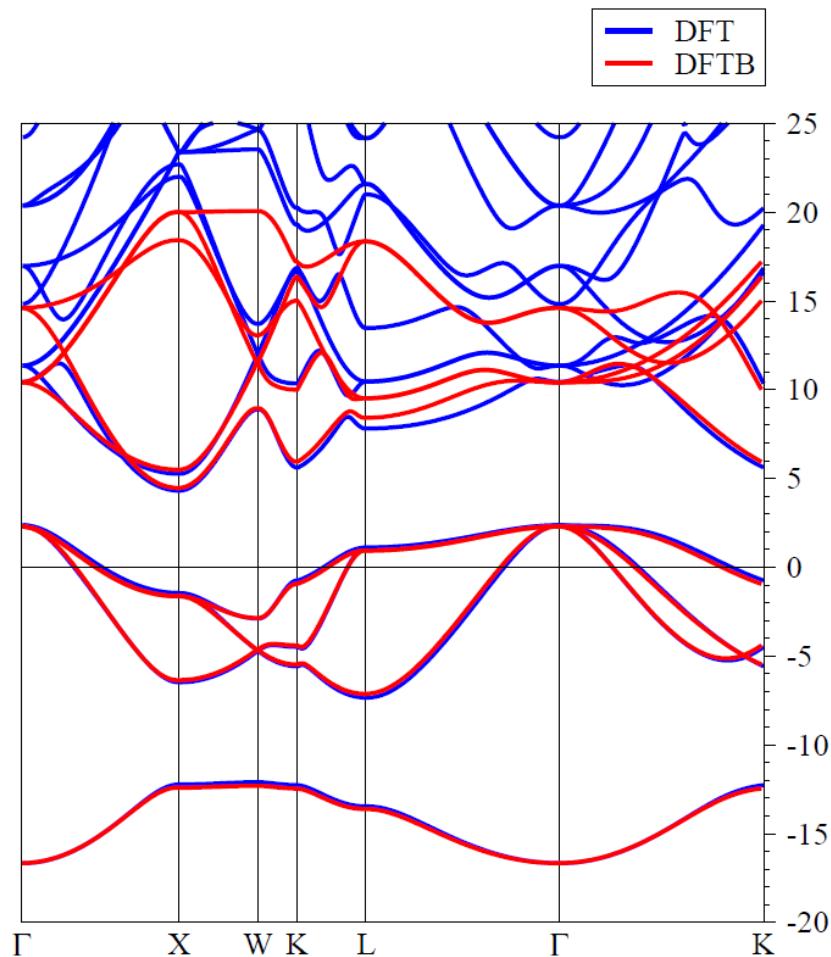
1). DFT band structure calculations

- VASP 4.6
- One atom per unit cell
- PAW (projector augmented wave) method
- $32 \times 32 \times 32$ Monkhorst-Pack k -point sampling
- cutoff = 400 eV
- Fermi level is shifted to 0 eV

2). DFTB band structure fitting

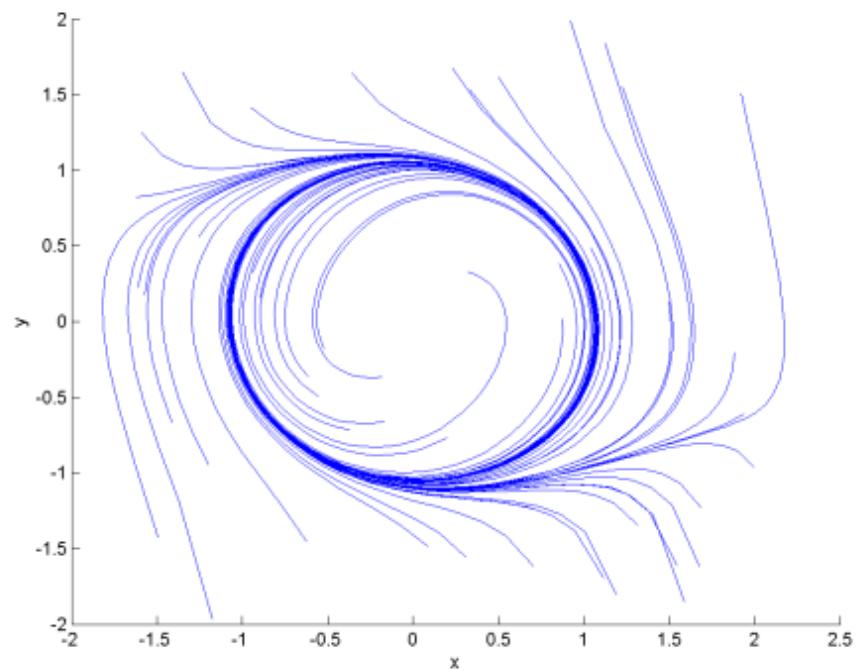
- Optimization of parameter sets for Woods-Saxon confining potential (orbital and density) and unoccupied orbital energies
- Fixed orbital energies for electron occupied orbitals
- Valence orbitals : [1s] for 1st row
[2s, 2p] for 2nd row
[ns, np, md] for 3rd – 6th row
($n \geq 3$, $m = n-1$ for group 1-12, $m = n$ for group 13-18)
- Fitting points : valence bands + conduction bands (depending on the system, at least including up to $\sim+5$ eV with respect to Fermi level)

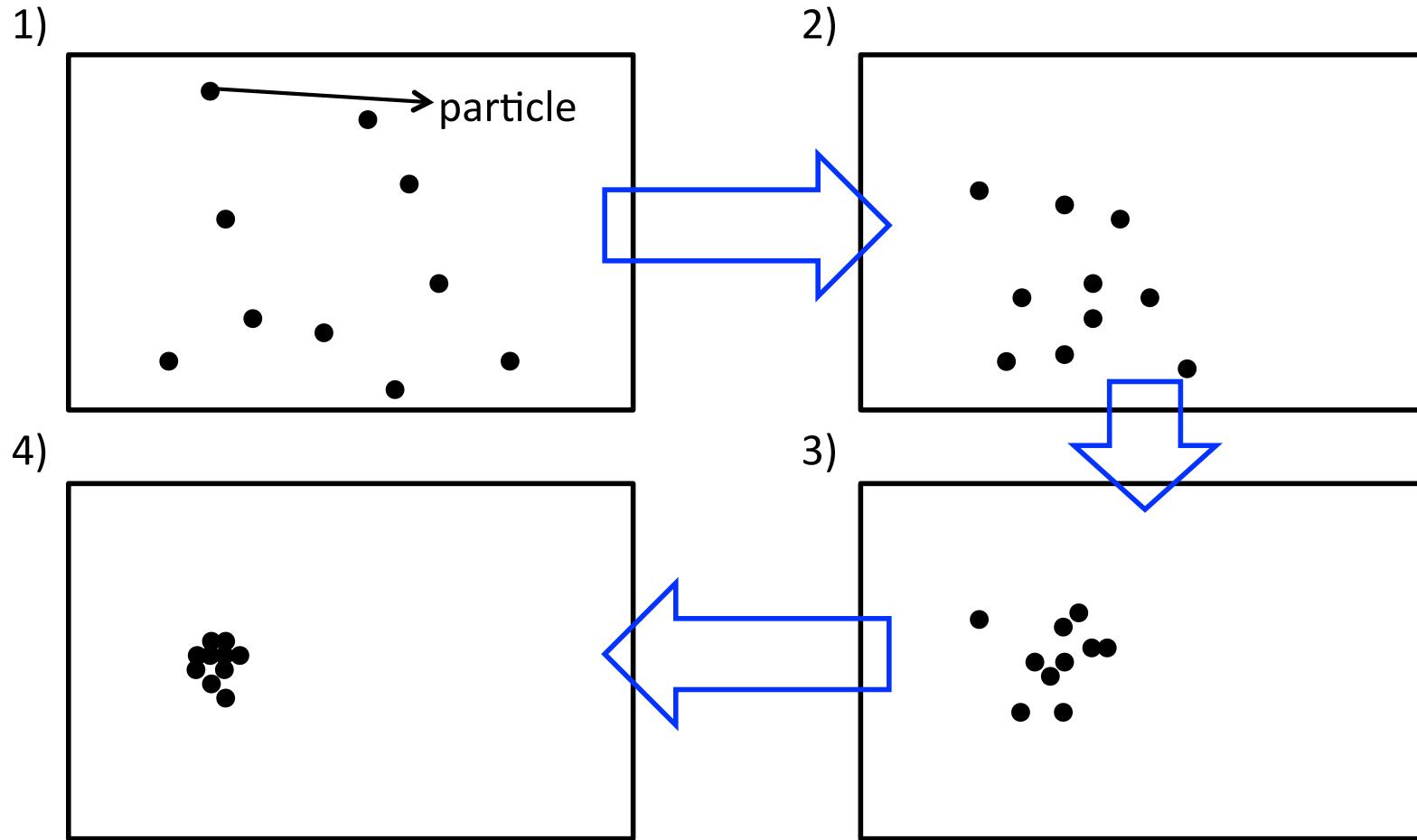
Band structure for Se (FCC)



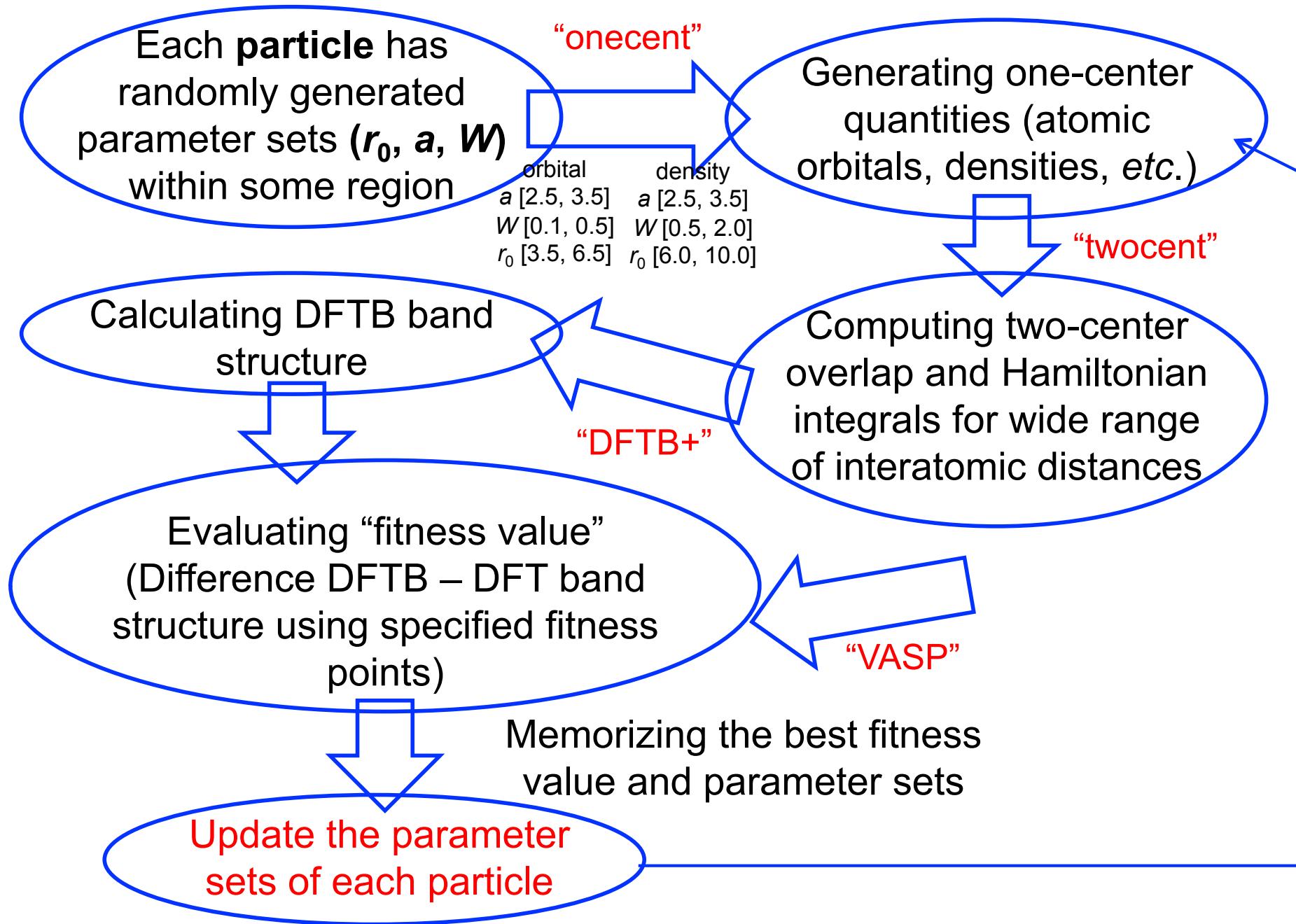
Brillouin zone

Particle swarm optimization (PSO)



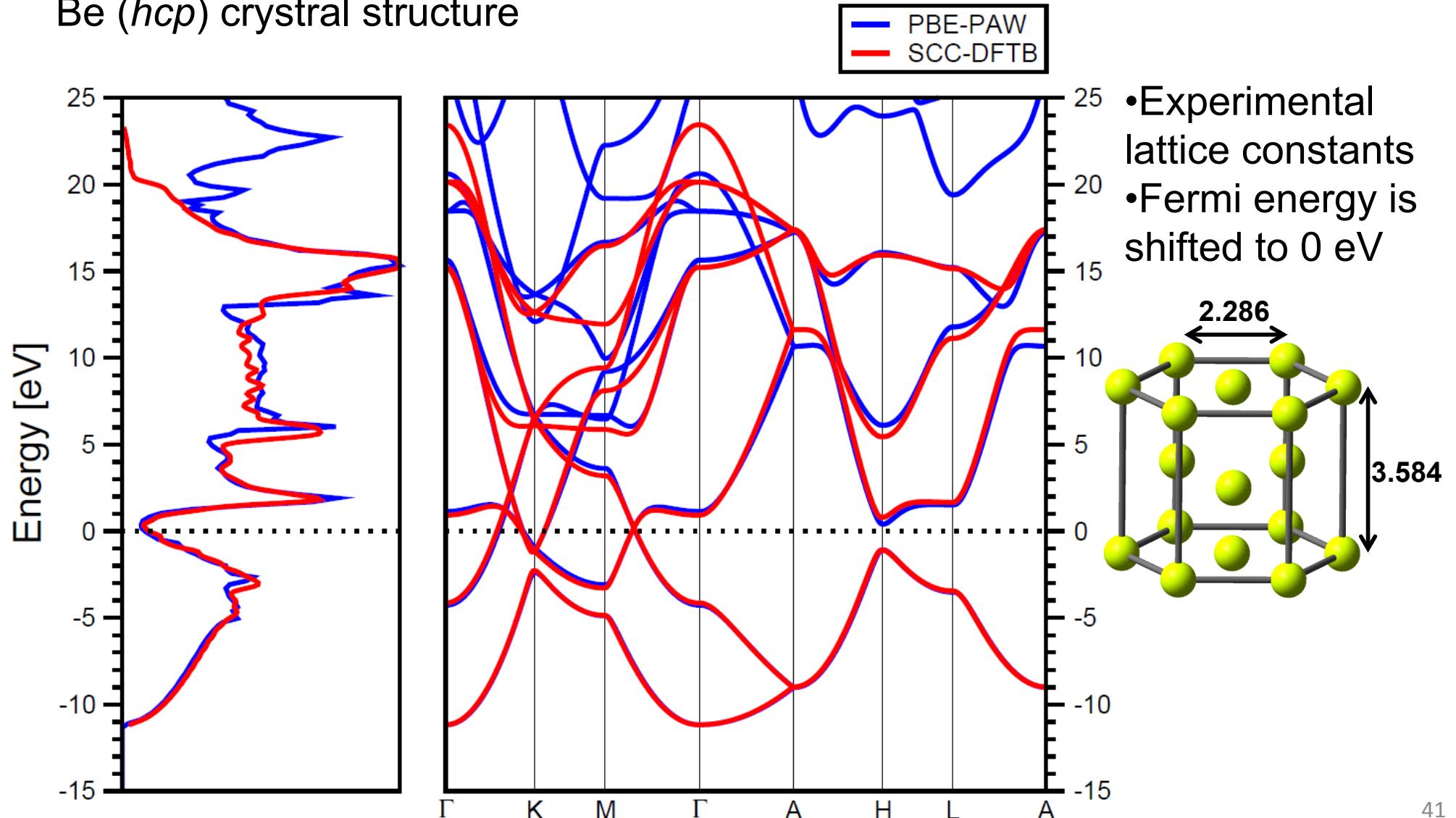


- 1) Particles (=candidate of a solution) are randomly placed initially in a target space.
- 2) – 3) Position and velocity of particles are updated based on the exchange of information between particles and particles try to find the best solution.
- 4) Particles converges to the place which gives the best solution after a number of iterations.

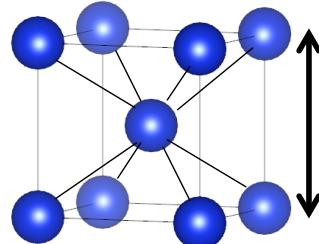


Example: Be, HCP crystal structure

Total density of states (left) and band structure (right) of Be (*hcp*) crystal structure



Band structure fitting for BCC crystal structures



- space group No. 229

- 1 lattice constant (a)

➤ No POTCAR file for $Z \geq 84$ in VASP

Group	1	2	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	18	
Period																			
1	1 H																2 He		
2		3 Li	4 Be																
3	11 Na	12 Mg																	
4	19 K	20 Ca	21 Sc	22 Ti	23 V	24 Cr	25 Mn	26 Fe	27 Co	28 Ni	29 Cu	30 Zn	13 Al	14 Si	15 P	16 S	17 Cl	18 Ar	
5	37 Rb	38 Sr	39 Y	40 Zr	41 Nb	42 Mo	43 Tc	44 Ru	45 Rh	46 Pd	47 Ag	48 Cd	31 In	32 Ga	33 Ge	34 As	35 Se	36 Br	
6	55 Cs	56 Ba	71 Lu	72 Hf	73 Ta	74 W	75 Re	76 Os	77 Ir	78 Pt	79 Au	80 Hg	81 Tl	82 Pb	83 Bi	84 Po	85 At	86 Rn	
7	87 Fr	88 Ra	*	103 Lr	104 Rf	105 Db	106 Sg	107 Bh	108 Hs	109 Mt	110 Ds	111 Rg	112 Cn	113 Uut	114 Uuq	115 Uup	116 Uuh	117 Uus	118 Uuo

— Transferability checked (single point calculation)

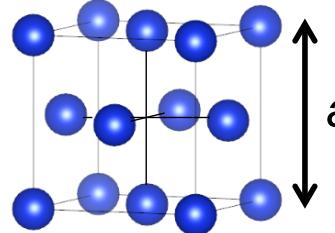
— Reference system in PSO

— Experimental lattice constants available

* Lanthanoids	*
** Actinoids	**

57 La	58 Ce	59 Pr	60 Nd	61 Pm	62 Sm	63 Eu	64 Gd	65 Tb	66 Dy	67 Ho	68 Er	69 Tm	70 Yb
89 Ac	90 Th	91 Pa	92 U	93 Np	94 Pu	95 Am	96 Cm	97 Bk	98 Cf	99 Es	100 Fm	101 Md	102 No

Band structure fitting for FCC crystal structures



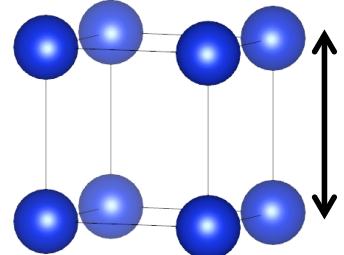
- space group No. 225

- 1 lattice constant (a)

Group	1	2	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	18	
Period																			
1	1 H																2 He		
2	3 Li	4 Be															10 Ne		
3	11 Na	12 Mg															18 Ar		
4	19 K	20 Ca	21 Sc	22 Ti	23 V	24 Cr	25 Mn	26 Fe	27 Co	28 Ni	29 Cu	30 Zn	31 Ga	32 Ge	33 As	34 Se	35 Br	36 Kr	
5	37 Rb	38 Sr	39 Y	40 Zr	41 Nb	42 Mo	43 Tc	44 Ru	45 Rh	46 Pd	47 Ag	48 Cd	49 In	50 Sn	51 Sb	52 Te	53 I	54 Xe	
6	55 Cs	56 Ba	*	71 Lu	72 Hf	73 Ta	74 W	75 Re	76 Os	77 Ir	78 Pt	79 Au	80 Hg	81 Tl	82 Pb	83 Bi	84 Po	85 At	86 Rn
7	87 Fr	88 Ra	**	103 Lr	104 Rf	105 Db	106 Sg	107 Bh	108 Hs	109 Mt	110 Ds	111 Rg	112 Cn	113 Uut	114 Uuq	115 Uup	116 Uuh	117 Uus	118 Uuo

* Lanthanoids	57 La	58 Ce	59 Pr	60 Nd	61 Pm	62 Sm	63 Eu	64 Gd	65 Tb	66 Dy	67 Ho	68 Er	69 Tm	70 Yb			
** Actinoids	89 Ac	90 Th	91 Pa	92 U	93 Np	94 Pu	95 Am	96 Cm	97 Bk	98 Cf	99 Es	100 Fm	101 Md	102 No			

Band structure fitting for SCL crystal structures



- space group No. 221

- 1 lattice constant (a)

Group	1	2	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	18
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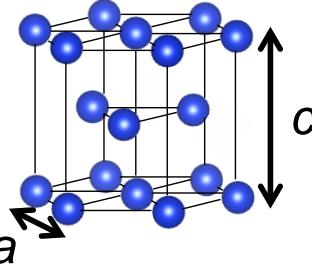
Period	1	2	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	18
--------	---	---	---	---	---	---	---	---	---	----	----	----	----	----	----	----	----	----

— Transferability checked (single point calculation)
 — Reference system in PSO
 — Experimental lattice constants available

1	H	2	He														
2	Li	Be															
3	Na	Mg															
4	K	Ca															
5	Rb	Sr															
6	Cs	Ba															
7	Fr	Ra															
8	Sc	Ti	V														
9	Zr	Nb	Mo														
10	Y	Tc															
11	Lu	Hf	Ta														
12	103	104	105														
13	106	107	108														
14	109	110	111														
15	112	113	114														
16	115	116	117														
17	118	119	120														
18	121	122	123														
*	21	22	23	24	25	26	27	28	29	30	31	32	33	34	35	36	37
**	71	72	73	74	75	76	77	78	79	80	81	82	83	84	85	86	87
***	103	104	105	106	107	108	109	110	111	112	113	114	115	116	117	118	119
****	106	107	108	109	110	111	112	113	114	115	116	117	118	119	120	121	122

* Lanthanoids	57	58	59	60	61	62	63	64	65	66	67	68	69	70	71	72	73
	La	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb			
** Actinoids	89	90	91	92	93	94	95	96	97	98	99	100	101	102	103	104	105
	Ac	Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No			

Band structure fitting for HCP crystal structures



- space group No. 194

- 2 lattice constants (a, c)

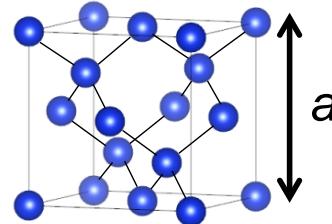
Group	1	2	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	18
Period																		
1	1 H																2 He	
2	3 Li	4 Be																
3	11 Na	12 Mg																
4	19 K	20 Ca	21 Sc	22 Ti	23 V	24 Cr	25 Mn	26 Fe	27 Co	28 Ni	29 Cu	30 Zn	31 Ga	32 Ge	33 As	34 Se	35 Br	36 Kr
5	37 Rb	38 Sr	39 Y	40 Zr	41 Nb	42 Mo	43 Tc	44 Ru	45 Rh	46 Pd	47 Ag	48 Cd	49 In	50 Sn	51 Sb	52 Te	53 I	54 Xe
6	55 Cs	56 Ba	71 Lu	72 Hf	73 Ta	74 W	75 Re	76 Os	77 Ir	78 Pt	79 Au	80 Hg	81 Tl	82 Pb	83 Bi	84 Po	85 At	86 Rn
7	87 Fr	88 Ra	103 Lr	104 Rf	105 Db	106 Sg	107 Bh	108 Hs	109 Mt	110 Ds	111 Rg	112 Cn	113 Uut	114 Uuq	115 Uup	116 Uuh	117 Uus	118 Uuo

— Transferability checked (single point calculation)
 — Reference system in PSO
 — Experimental lattice constants available

* Lanthanoids
** Actinoids

*	57 La	58 Ce	59 Pr	60 Nd	61 Pm	62 Sm	63 Eu	64 Gd	65 Tb	66 Dy	67 Ho	68 Er	69 Tm	70 Yb
**	89 Ac	90 Th	91 Pa	92 U	93 Np	94 Pu	95 Am	96 Cm	97 Bk	98 Cf	99 Es	100 Fm	101 Md	102 No

Band structure fitting for Diamond crystal structures



- space group No. 227

- 1 lattice constant (a)

Group	1	2	3	4	5	6	7	8	9	10	11	12	13	14	15	16	17	18	
Period																			
1	1 H																	2 He	
2	3 Li	4 Be																	
3	11 Na	12 Mg																	
4	19 K	20 Ca	21 Sc	22 Ti	23 V	24 Cr	25 Mn	26 Fe	27 Co	28 Ni	29 Cu	30 Zn	31 Ga	32 Ge	33 As	34 Se	35 Br	36 Kr	
5	37 Rb	38 Sr	39 Y	40 Zr	41 Nb	42 Mo	43 Tc	44 Ru	45 Rh	46 Pd	47 Ag	48 Cd	49 In	50 Sn	51 Sb	52 Te	53 I	54 Xe	
6	55 Cs	56 Ba	*	71 Lu	72 Hf	73 Ta	74 W	75 Re	76 Os	77 Ir	78 Pt	79 Au	80 Hg	81 Tl	82 Pb	83 Bi	84 Po	85 At	86 Rn
7	87 Fr	88 Ra	**	103 Lr	104 Rf	105 Db	106 Sg	107 Bh	108 Hs	109 Mt	110 Ds	111 Rg	112 Cn	113 Uut	114 Uuq	115 Uup	116 Uuh	117 Uus	118 Uuo

— Transferability checked (single point calculation)

— Reference system in PSO

— Experimental lattice constants available

* Lanthanoids	57 La	58 Ce	59 Pr	60 Nd	61 Pm	62 Sm	63 Eu	64 Gd	65 Tb	66 Dy	67 Ho	68 Er	69 Tm	70 Yb
** Actinoids	89 Ac	90 Th	91 Pa	92 U	93 Np	94 Pu	95 Am	96 Cm	97 Bk	98 Cf	99 Es	100 Fm	101 Md	102 No

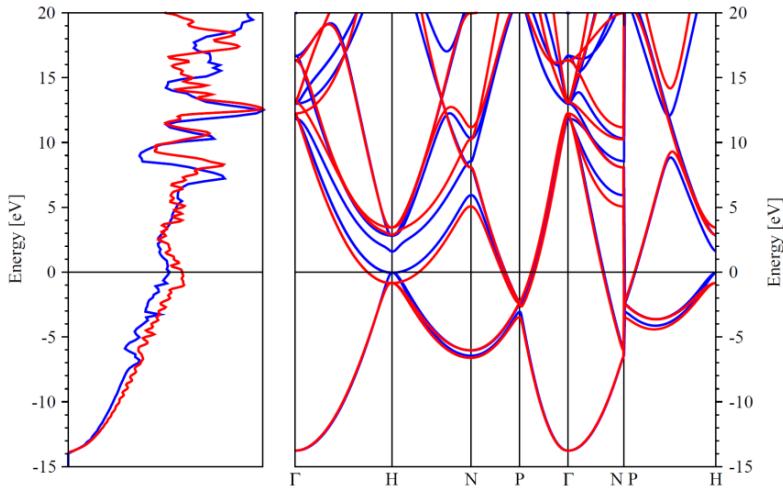
Transferability of optimum parameter sets for different structures

DFTB Parameterization

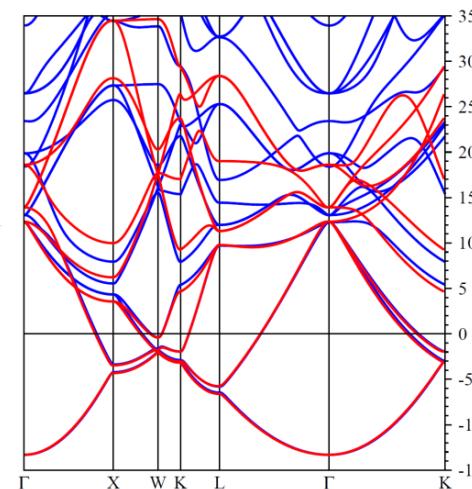
— DFT
— DFTB

e.g. : Si, parameters were optimized with bcc only

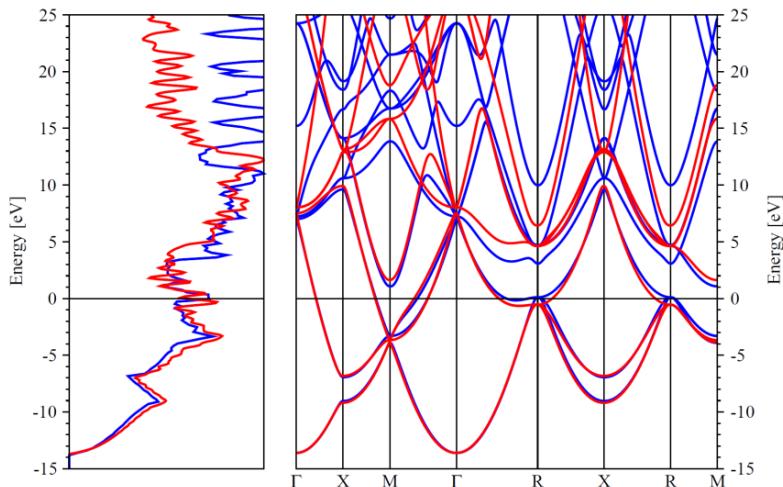
bcc



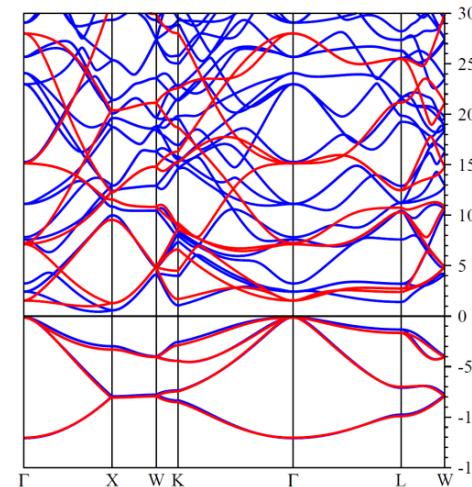
fcc



scl



diamond



$3s^23p^23d^0$

Lattice constants:

Expt

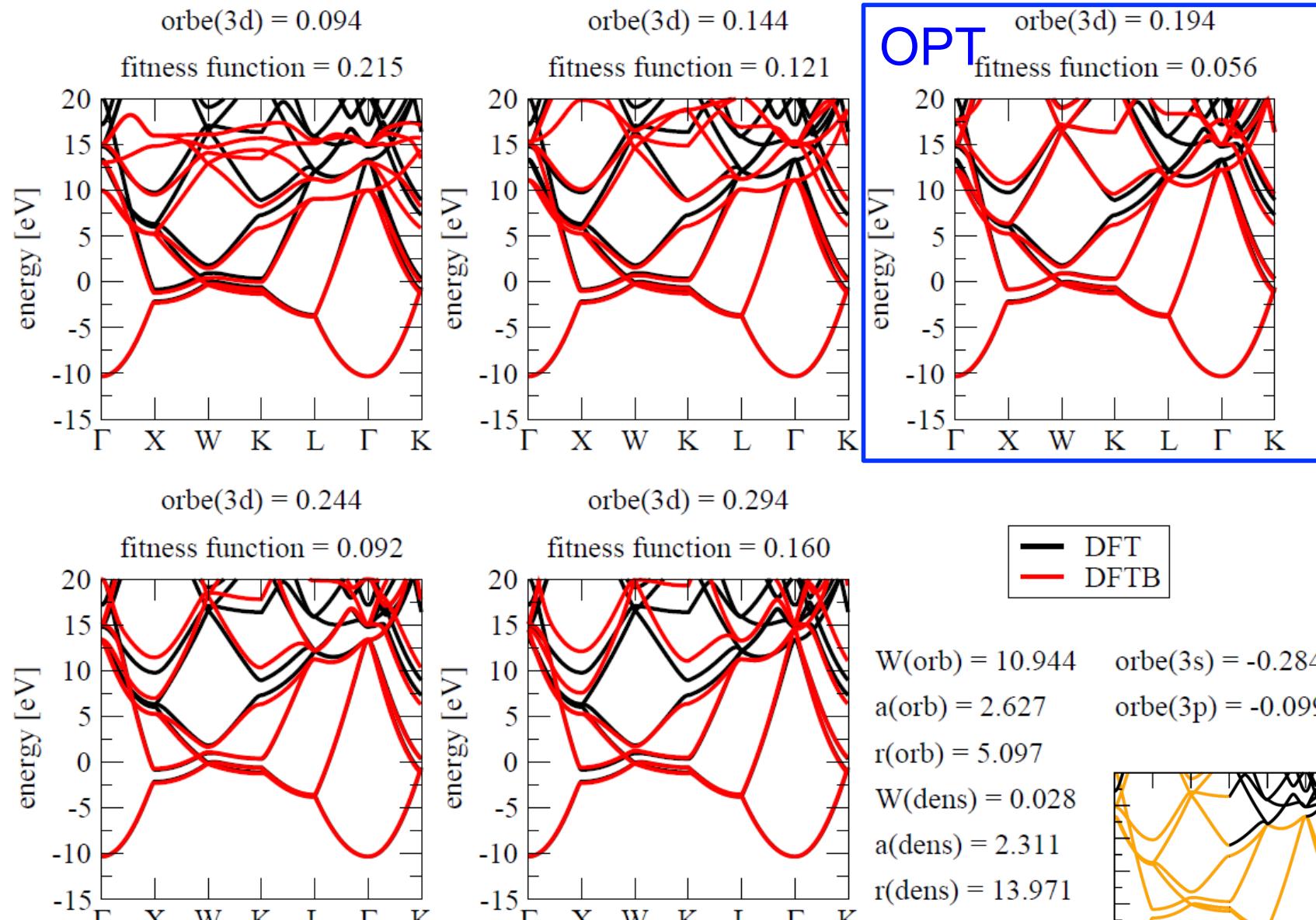
· bcc	3.081
fcc	3.868
scl	2.532
diamond	5.431

Parameter sets:

W (orb)	3.33938
a (orb)	4.52314
r (orb)	4.22512
W (dens)	1.68162
a (dens)	2.55174
r (dens)	9.96376
ε_s	-0.39735
ε_p	-0.14998
ε_d	0.21210

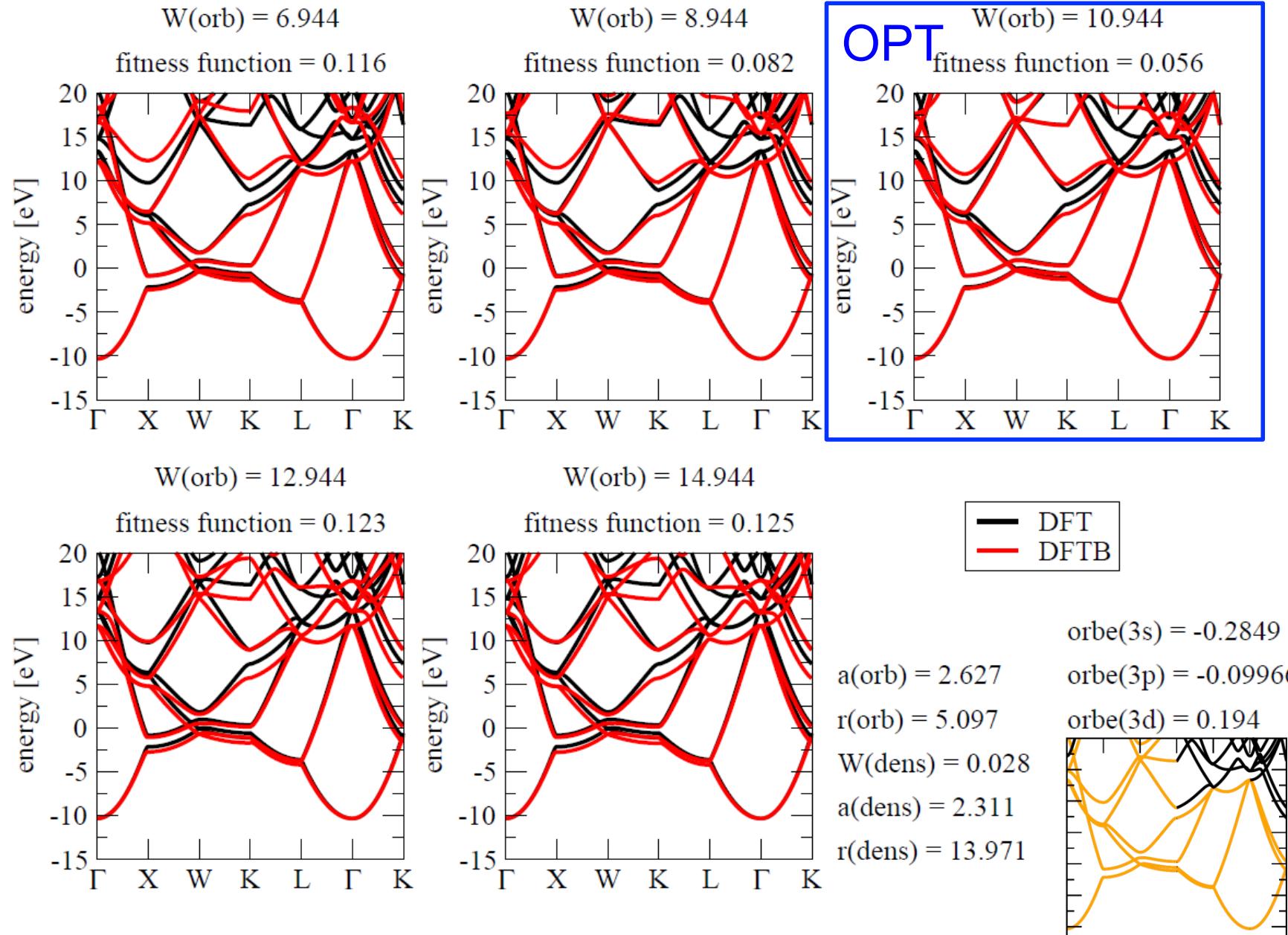
➤ Artificial crystal structures can be reproduced well

Influence of virtual orbital energy (3d) to Al (fcc) band structure



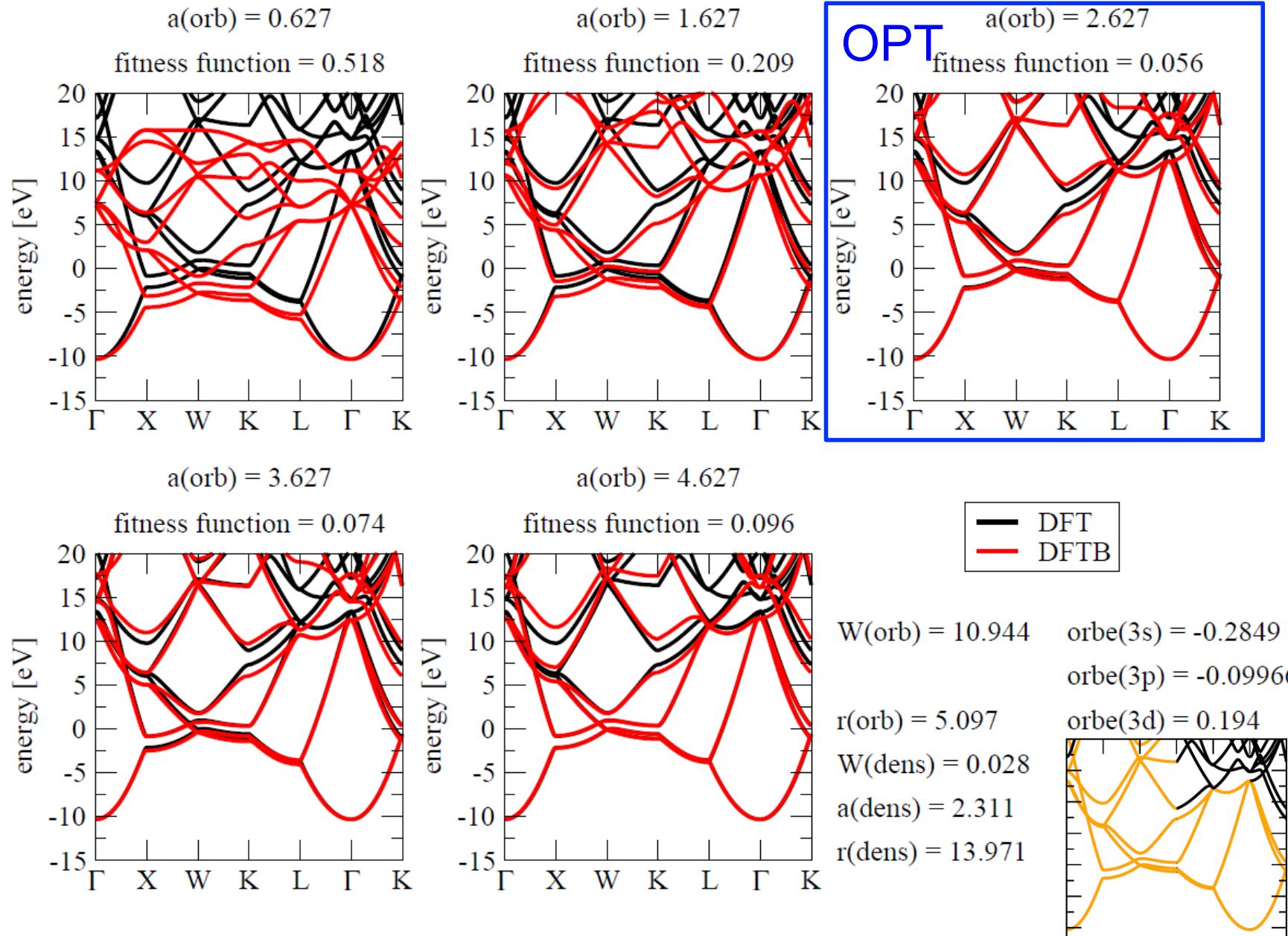
➤ The bands of upper part are shifted up constantly as $\text{orb}\varepsilon(3d)$ becomes larger

Influence of $W(\text{orb})$ to Al (*fcc*) band structure



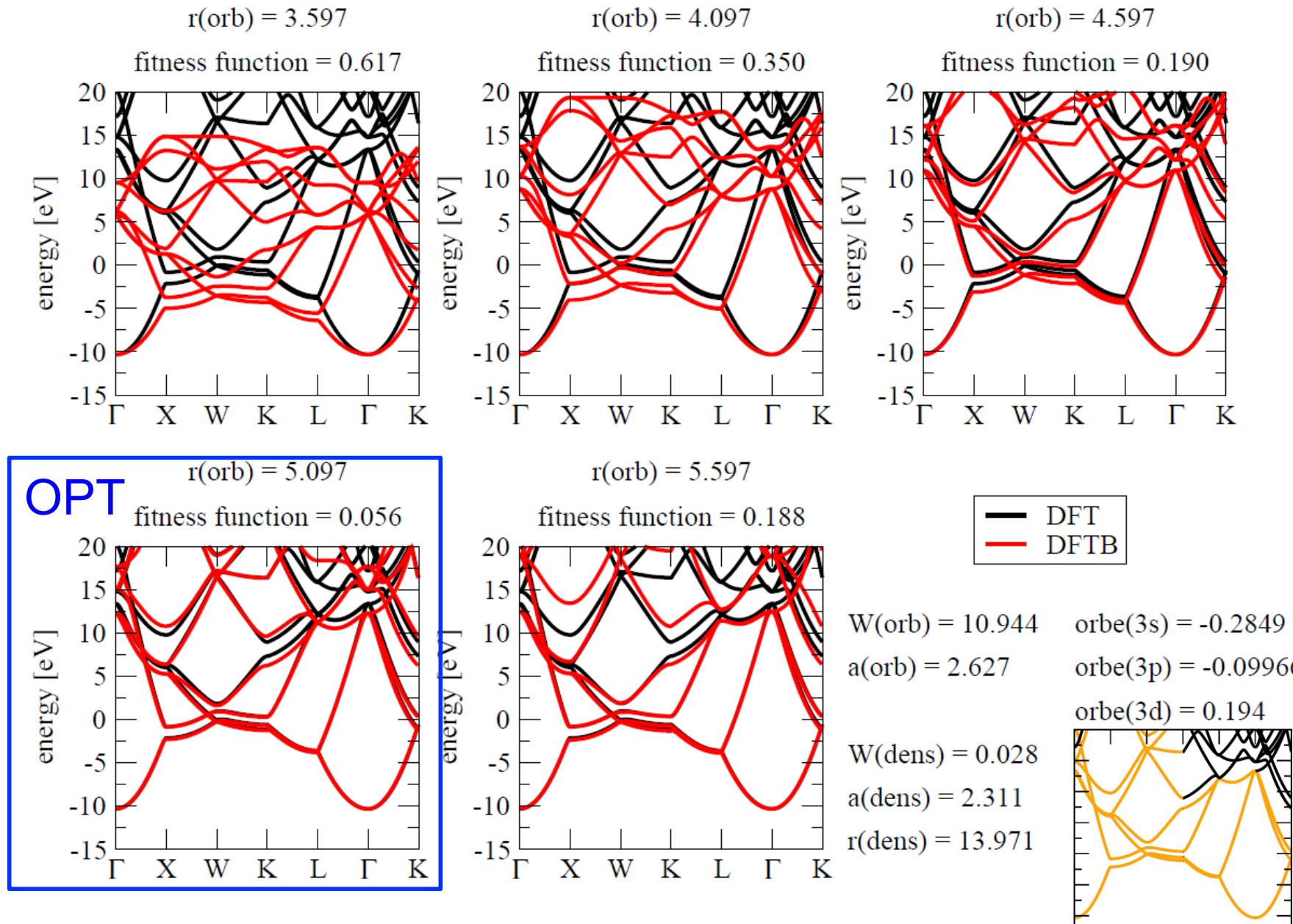
➤ The bands of upper part go lower as $W(\text{orb})$ becomes larger

Influence of $a(\text{orb})$ to Al (*fcc*) band structure

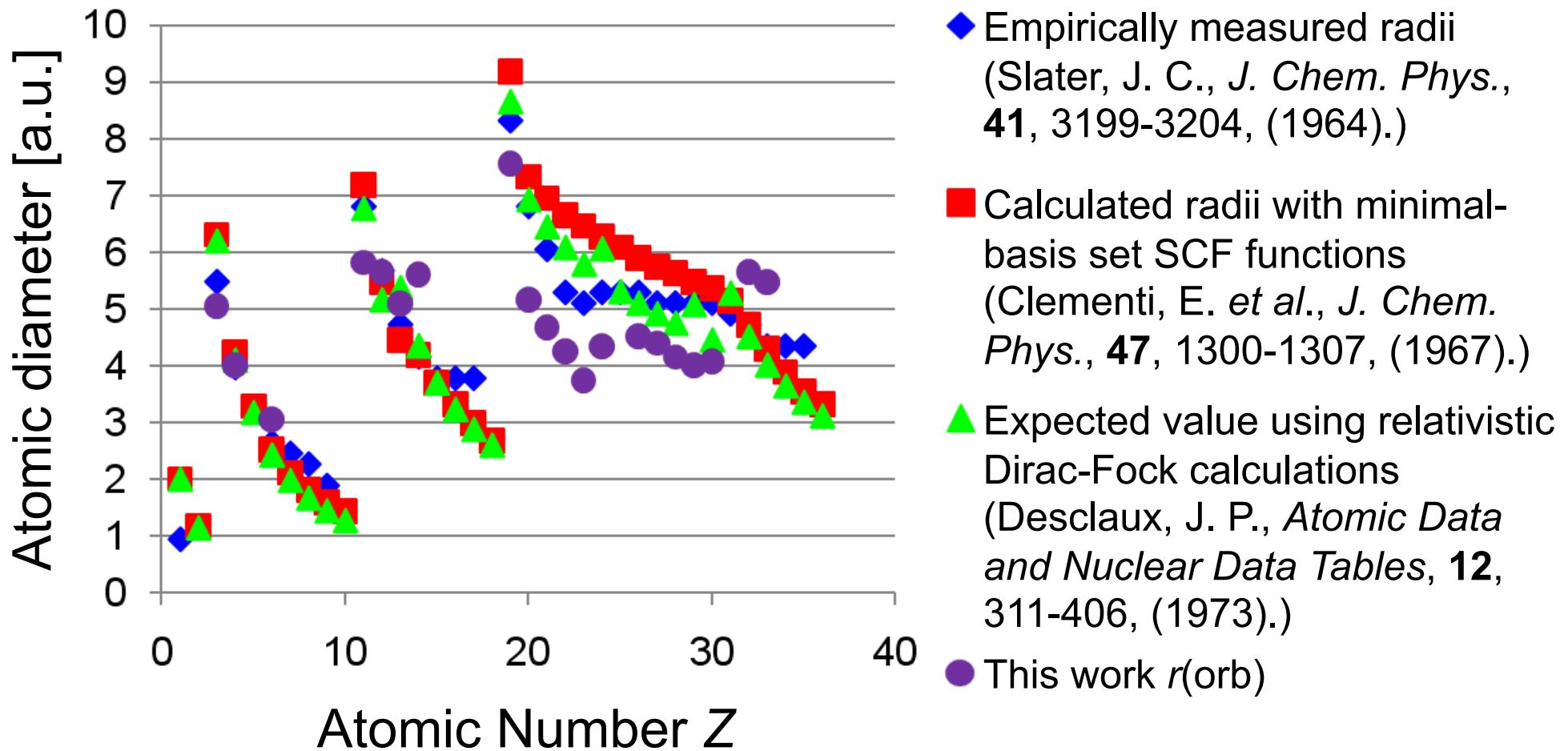


➤ Too small $a(\text{orb})$ gives the worse band structure

Influence of $r(\text{orb})$ to Al (fcc) band structure



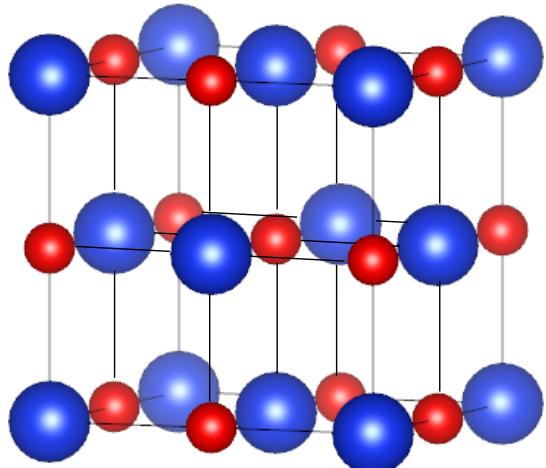
➤ $r(\text{orb})$ strongly influences DFTB band structure

Correlation of $r(\text{orb})$ vs. atomic diameter

➤ In particular for main group elements, there seems to be a correlation between $r(\text{orb})$ and atomic diameter.

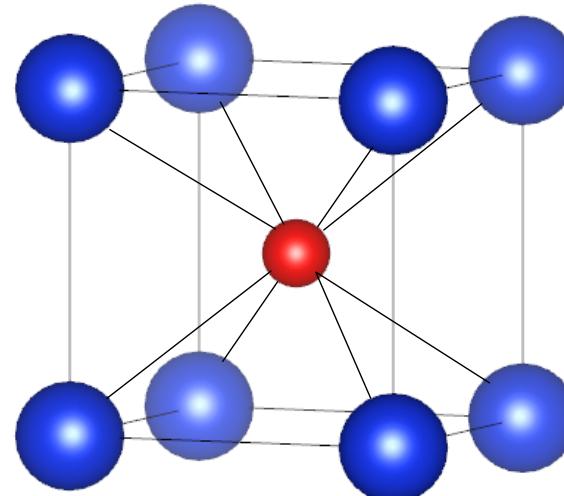
Straightforward application to binary crystal structures

Rocksalt (space group No. 225)



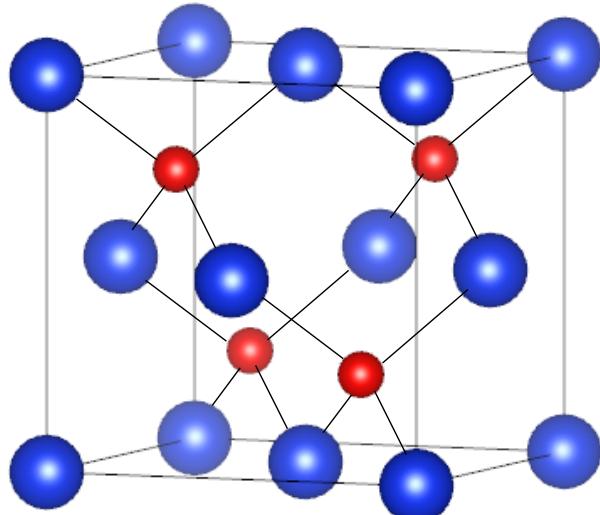
- NaCl
- MgO
- MoC
- AgCl
- ...

B2 (space group No. 221)



- CsCl
- FeAl
- ...

Zincblende (space group No. 216)



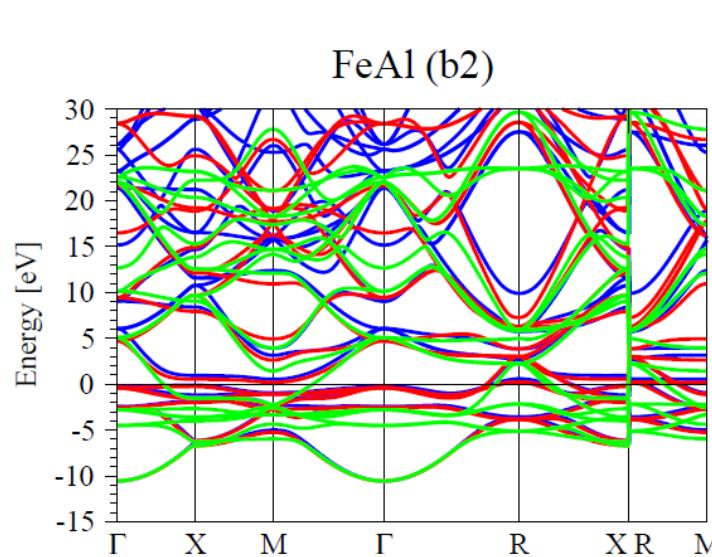
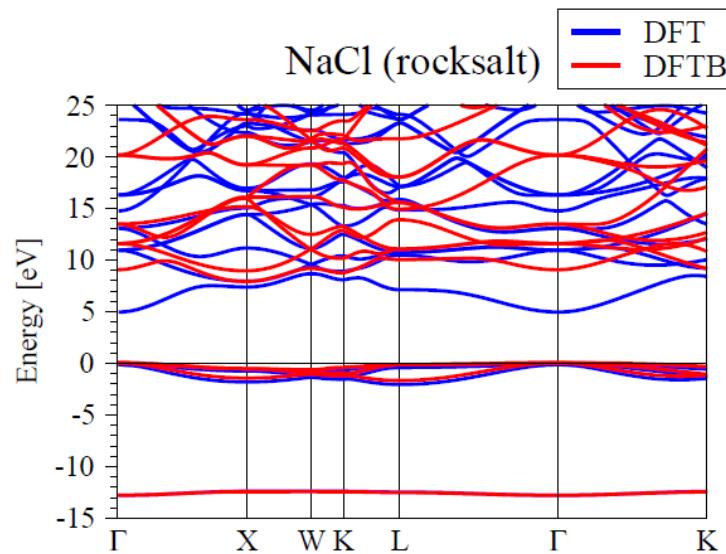
- SiC
- CuCl
- ZnS
- GaAs
- ...

Others

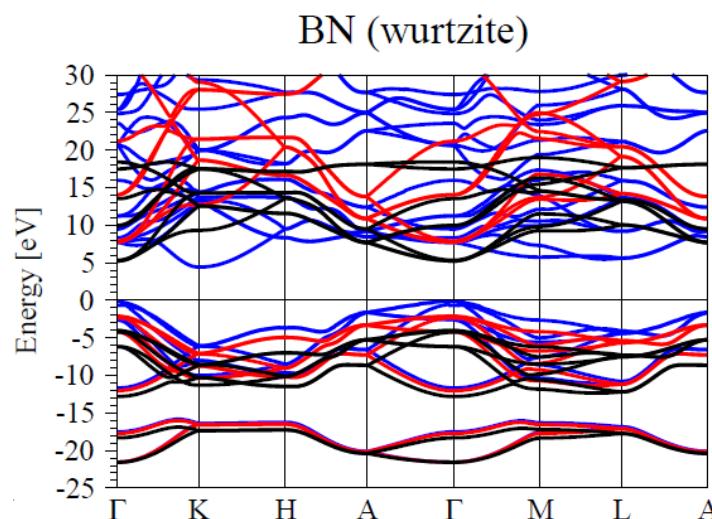
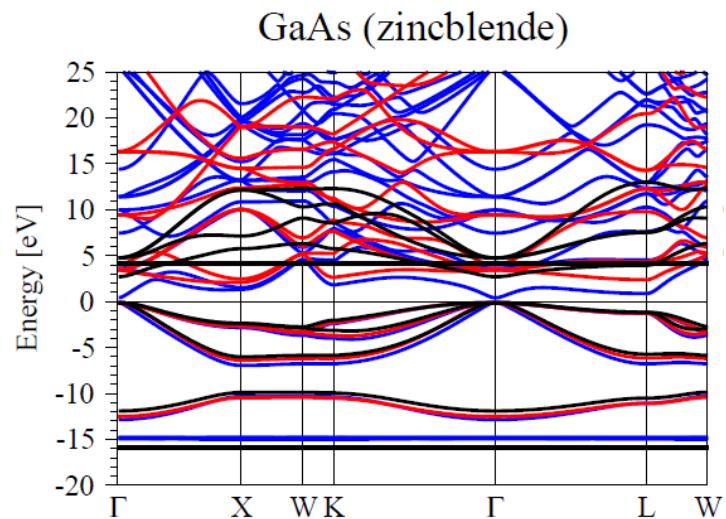
- Wurtzite (BeO, AlO, ZnO, GaN, ...)
- Hexagonal (BN, WC)
- Rhombohedral (*ABCABC* stacking sequence, BN)

➤ more than 100 pairs tested

Selected examples for binary crystal structures



- $d^7 s^1$ is used in POTCAR (DFT)

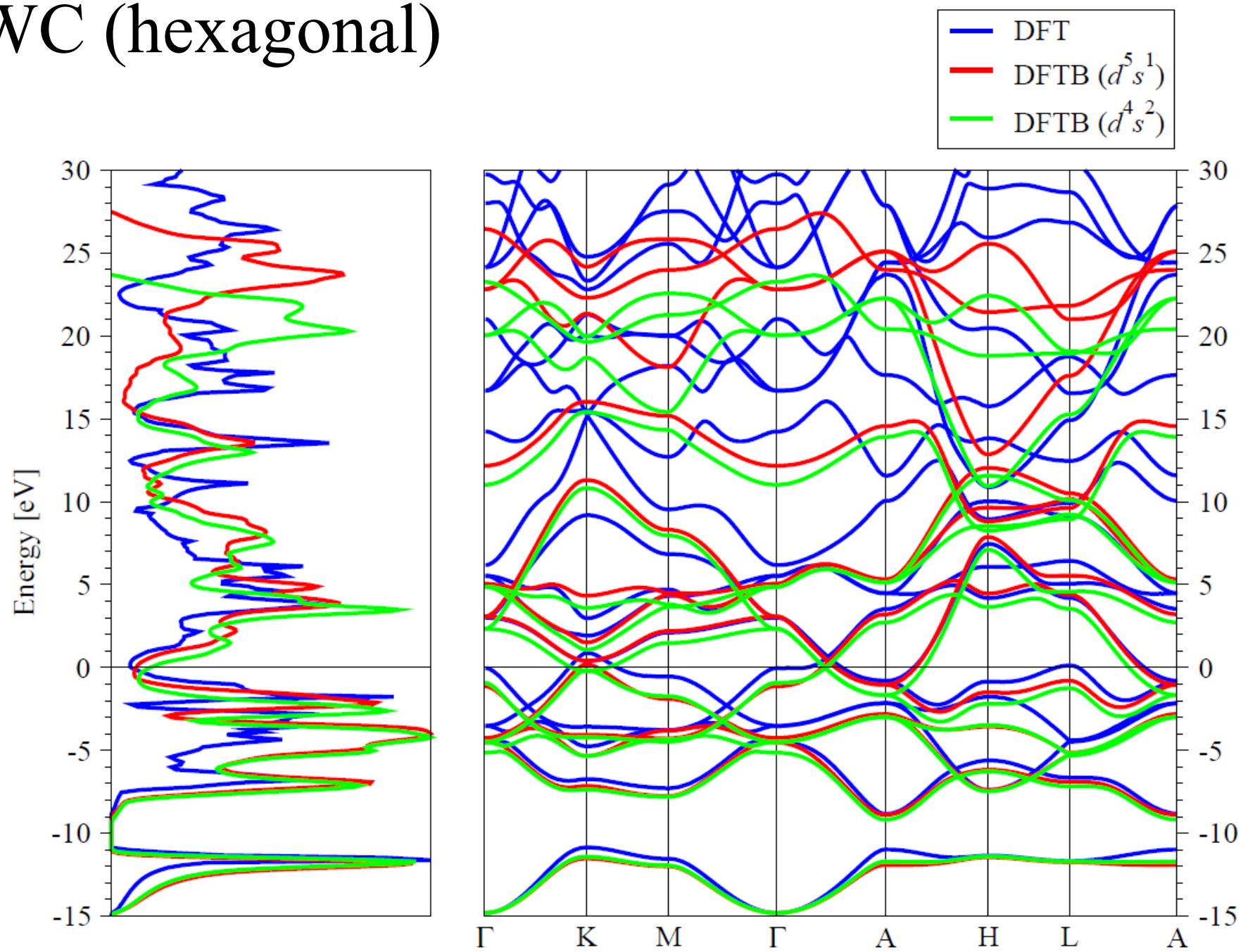


Reference of previous work :

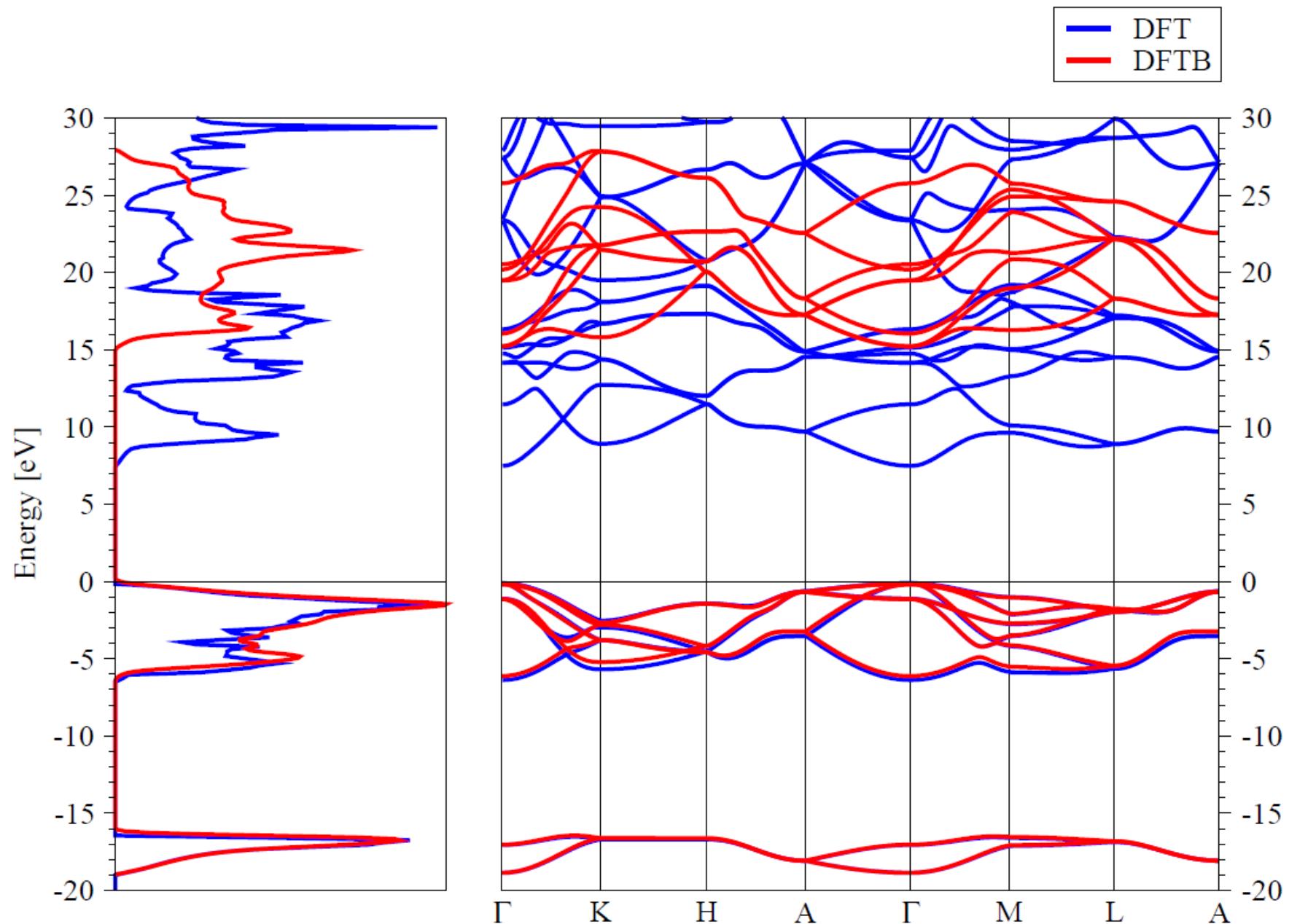
element	name
Ga, As	hyb-0-2
B, N	matsci-0-2

➤ Further improvement can be performed for specific purpose but this preliminary sets will work as good starting points

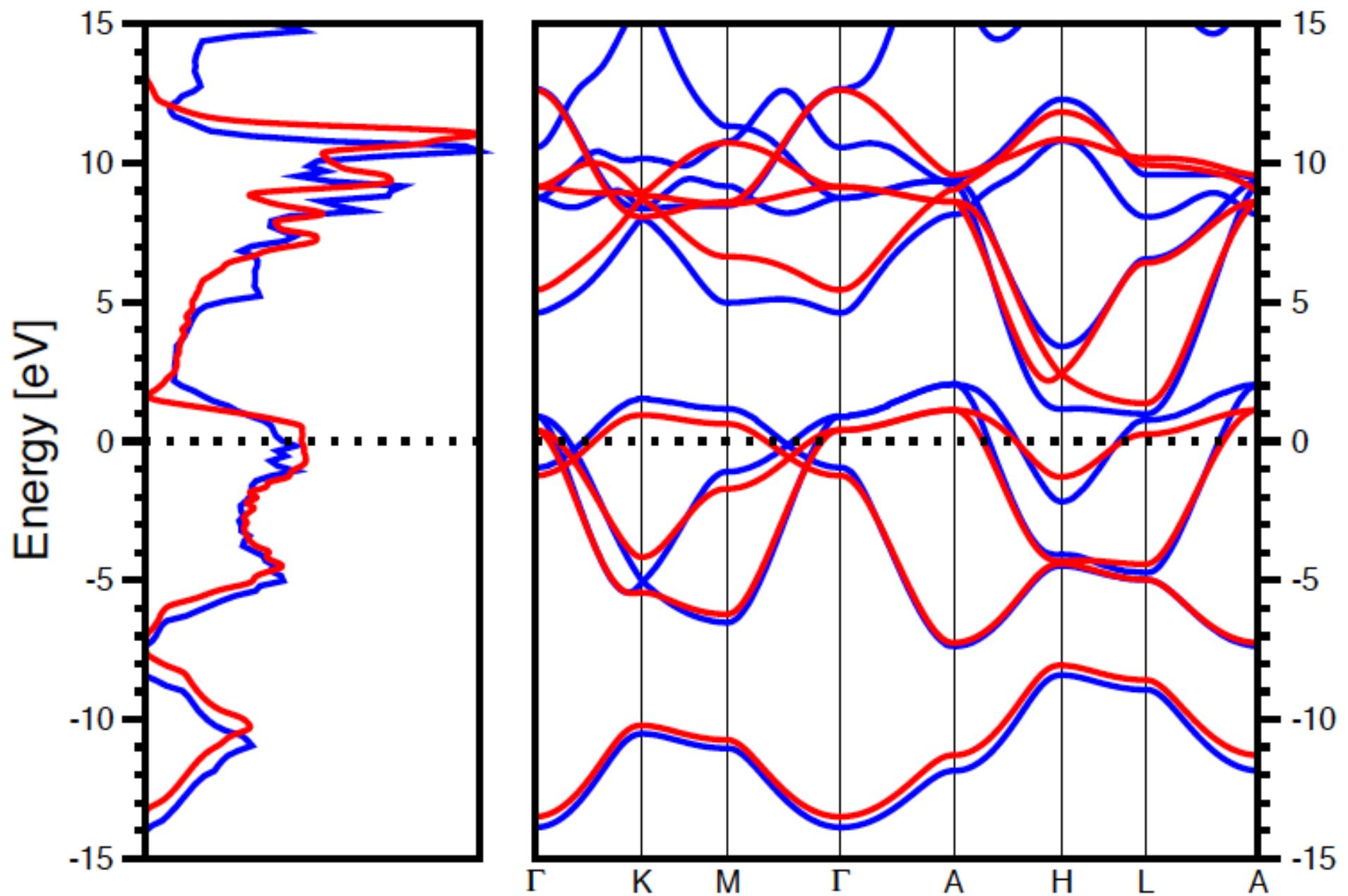
WC (hexagonal)



BeO

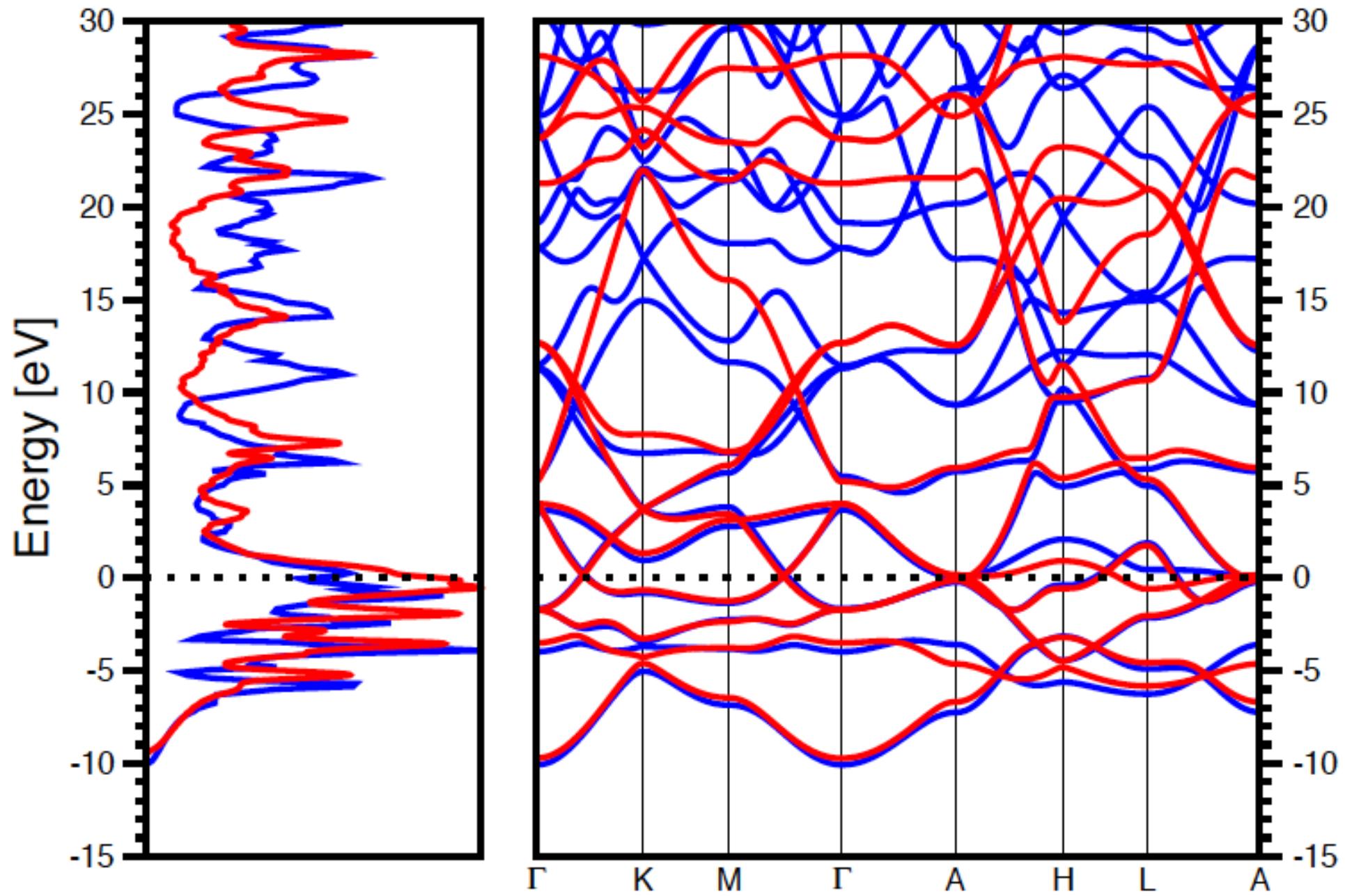


BeC



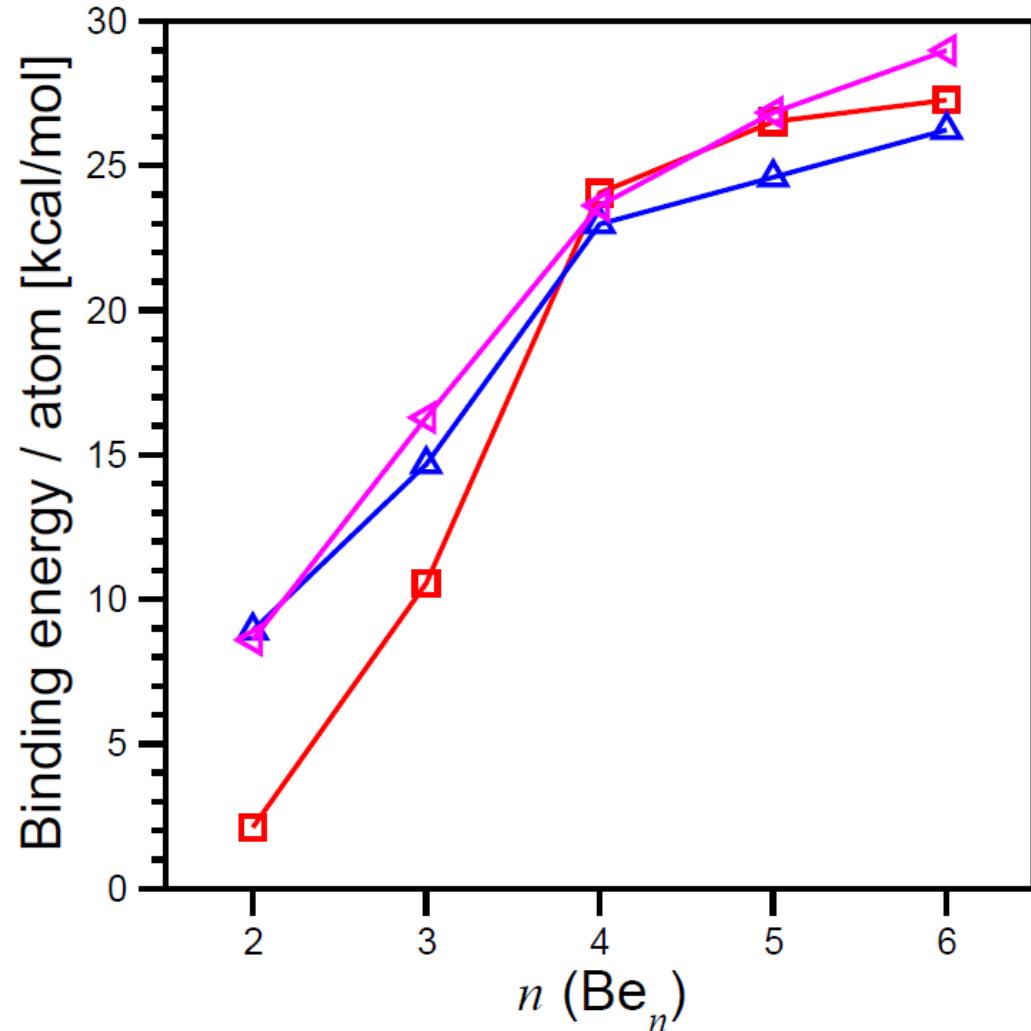
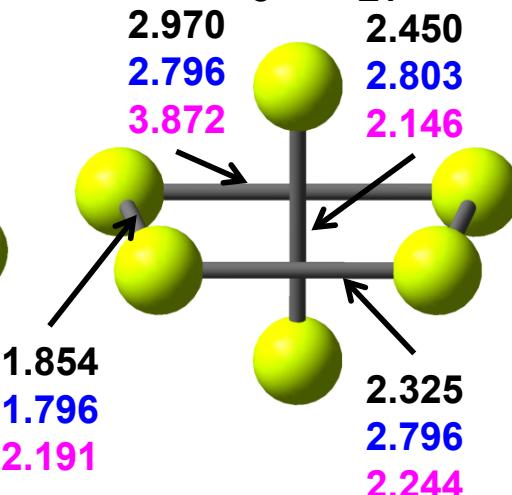
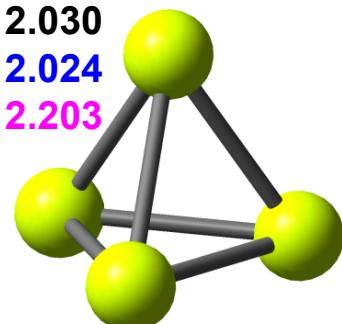
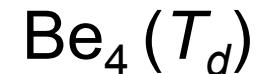
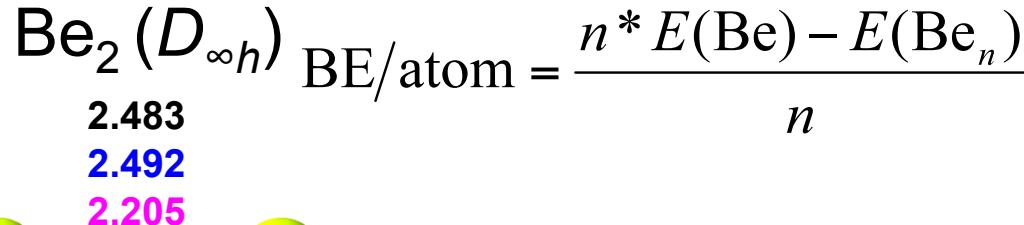
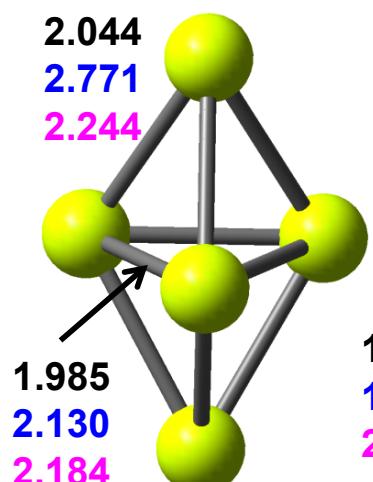
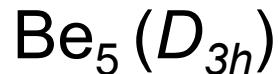
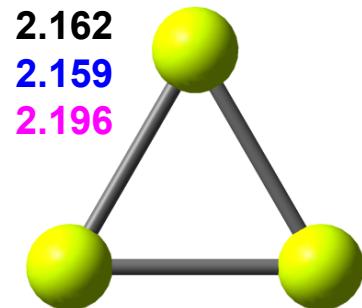
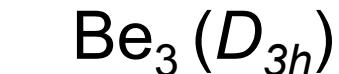
WBe

PBE-PAW
SCC-DFTB



Description of small neutral beryllium clusters

- Distance in [Å]
 - Closed shell singlet
- B3LYP/cc-pVTZ**
SCC-DFTB
PM3



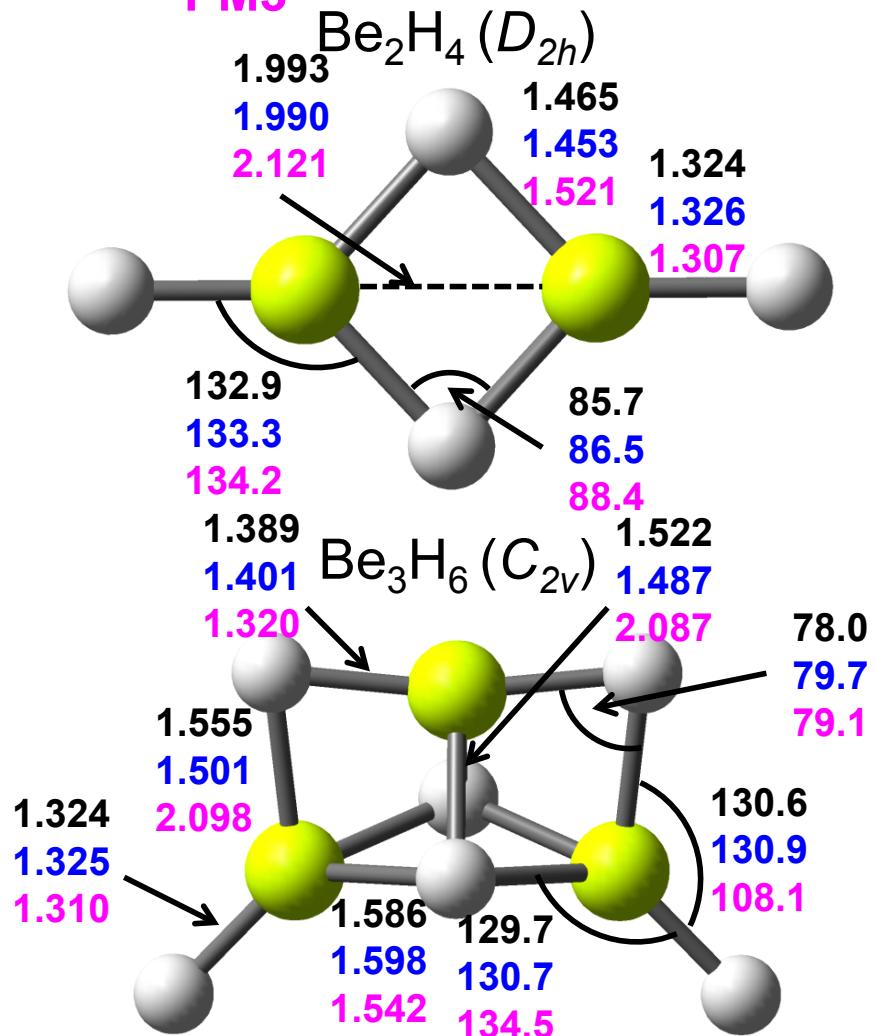
Geometries of beryllium hydride molecules

- Distance in [Å]
- Angle in [degree]
- Closed shell singlet

B3LYP/cc-pVTZ

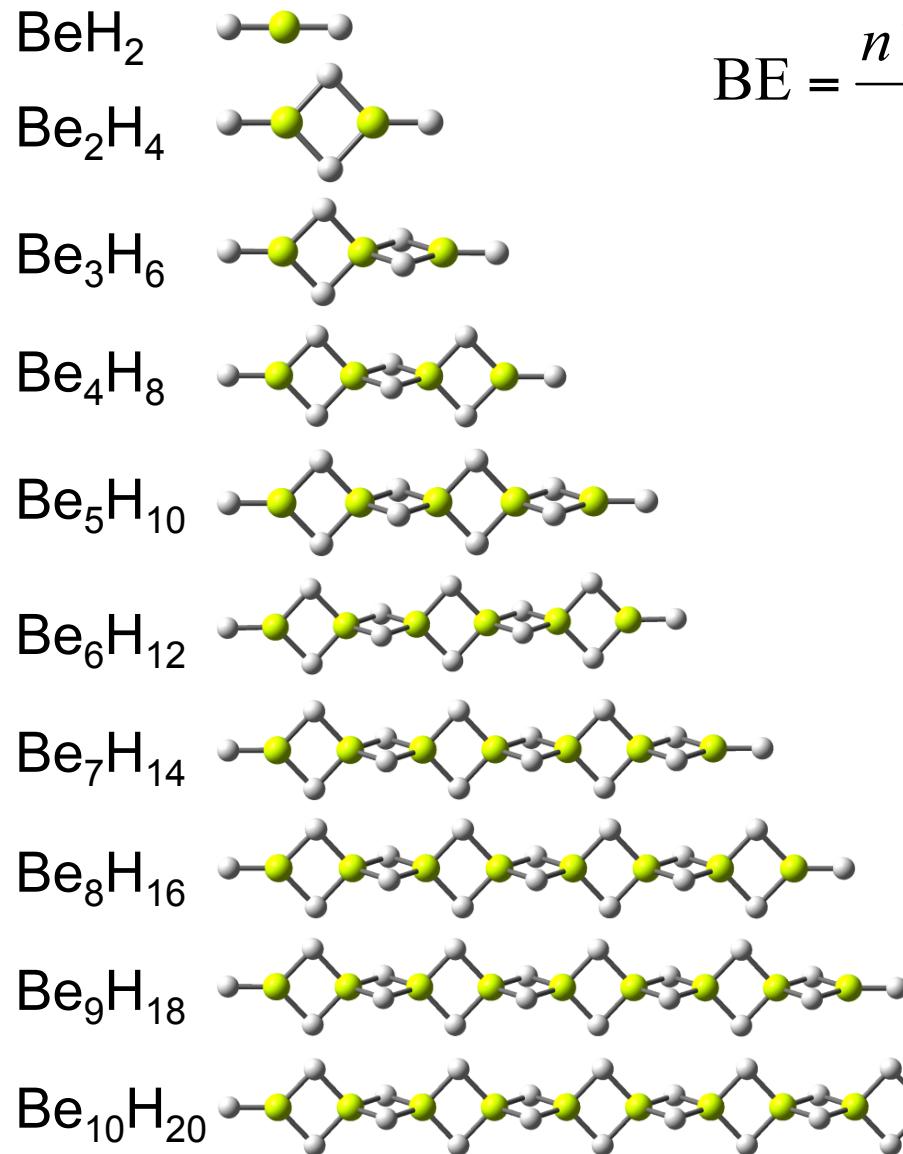
SCC-DFTB

PM3

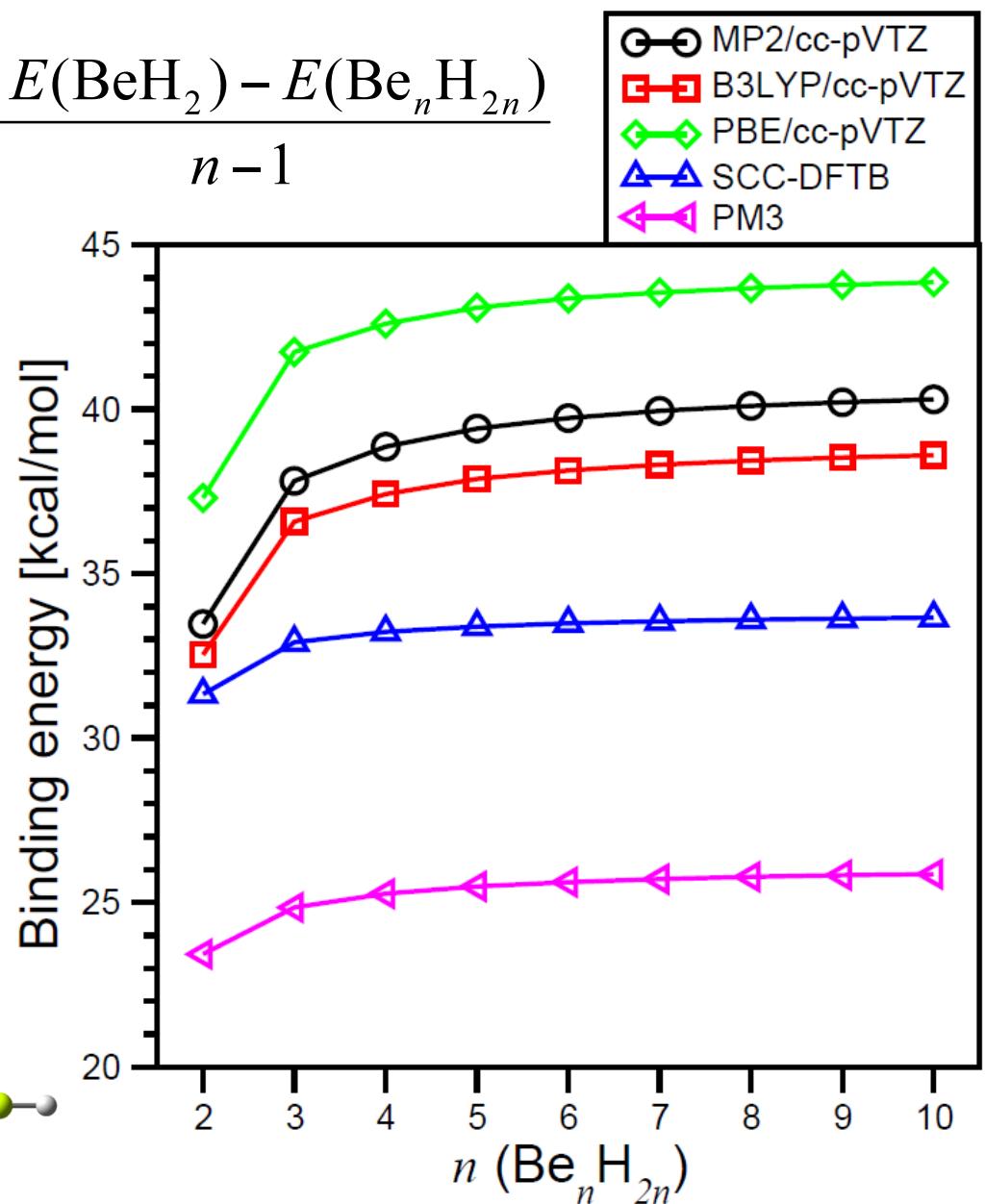


➤ DFTB potentially predicts good geometries for small beryllium molecules

Binding energy of beryllium hydride oligomers



$$BE = \frac{n * E(BeH_2) - E(Be_n H_{2n})}{n - 1}$$



➤ Discrepancy of DFTB binding energy from DFT or MP2 is reduced compared with other semi-empirical methods (PM3)

- Approximate DFT and DFT methods useful to study quantum chemistry of complex systems
- DFTB Be-X parameters will be used for:
 - A) chemical sputtering simulations for calibration/validation of classical potential simulations
 - B) structure/energy calculations of amorphous systems

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Thank you for your attention!