Quantum Espresso: Materials Modeling with Density Functional Theory

Davide Donadio, UC Davis

Giannozzi, P.; Andreussi, O.; Brumme, T.; Bunau, O.; Buongiorno Nardelli, M.; Calandra, M.; Car, R.; Cavazzoni, C.; Ceresoli, D.; Cococcioni, M.; Colonna, N.; Carnimeo, I.; Dal Corso, A.; de Gironcoli, S.; Delugas, P.; DiStasio, R. A.; Ferretti, A.; Floris, A.; Fratesi, G.; Fugallo, G.; Gebauer, R.; Gerstmann, U.; Giustino, F.; Gorni, T.; Jia, J.; Kawamura, M.; Ko, H.-Y.; Kokalj, A.; Küçükbenli, E.; Lazzeri, M.; Marsili, M.; Marzari, N.; Mauri, F.; Nguyen, N. L.; Nguyen, H.-V.; Otero-de-la-Roza, A.; Paulatto, L.; Poncé, S.; Rocca, D.; Sabatini, R.; Santra, B.; Schlipf, M.; Seitsonen, A. P.; Smogunov, A.; Timrov, I.; Thonhauser, T.; Umari, P.; Vast, N.; Wu, X.; Baroni, S. **Advanced Capabilities for Materials Modelling with Quantum ESPRESSO**. *J. Phys.: Condens. Matter* **2017**, *29* (46), 465901. https://doi.org/10.1088/1361-648X/aa8f79.

Giannozzi, P.; Baroni, S.; Bonini, N.; Calandra, M.; Car, R.; Cavazzoni, C.; Ceresoli, D.; Chiarotti, G. L.; Cococcioni, M.; Dabo, I.; Dal Corso, A.; de Gironcoli, S.; Fabris, S.; Fratesi, G.; Gebauer, R.; Gerstmann, U.; Gougoussis, C.; Kokalj, A.; Lazzeri, M.; Martin-Samos, L.; Marzari, N.; Mauri, F.; Mazzarello, R.; Paolini, S.; Pasquarello, A.; Paulatto, L.; Sbraccia, C.; Scandolo, S.; Sclauzero, G.; Seitsonen, A. P.; Smogunov, A.; Umari, P.; Wentzcovitch, R. M. QUANTUM ESPRESSO: A Modular and Open-Source Software Project for Quantum Simulations of Materials. *J. Phys.: Condens. Matter* 2009, 21 (39), 395502. https://doi.org/10.1088/0953-8984/21/39/395502.

Materials Modeling by First Principles Goals:

- Compute materials properties using only basic physics principles and their chemical composition (+crystal structure)
- Accuracy and predictivity, without relying on empirical parameters (e.g. from experiments)
- Efficiency, without compromising scientific rigor

Materials properties

- Ground state equilibrium properties:
 - Density, structural parameters, formation energy
- Nonequilibrium and excited state properties:
 - Mechanical response, vibrational response (phonons, IR and Raman spectra), chemical shift (NMR)
 - Transport coefficients (charge and thermal transport)
 - UV-visible and X-ray spectroscopy, photoemission

The many-body Schrödinger equation for electrons and nuclei

$$i\hbar \frac{\partial \Psi(r,R,t)}{\partial t} = \hat{H}\Psi(R,r,t)$$

$$\hat{H} = -\frac{\hbar^2}{2M}\nabla_R^2 - \frac{\hbar^2}{2m}\nabla_r^2 + V(r,R)$$

The many-body Schrödinger equation for electrons and nuclei

$$i\hbar \frac{\partial \Psi(r,R,t)}{\partial t} = \hat{H}\Psi(R,r,t)$$

$$\hat{H} = -\frac{\hbar^2}{2M}\nabla_r^2 - \frac{\hbar^2}{2m}\nabla_r^2 + V(r,R)$$

Born-Oppenheimer (adiabatic) approximation: $m \ll M$

The many-body Schrödinger equation for electrons and nuclei

electrons:
$$\left(-\frac{\hbar^2}{2m} \nabla_r^2 + V_R(r) \right) \psi_R(r) = E(r, R) \psi_R(r)$$

nuclei:

$$M\frac{d^2R}{dt^2} = -\nabla_R E(r,R)$$

The many-body Hamiltonian for the electrons is:

$$\hat{H} = -\frac{\hbar^2}{2m} \sum_{i} \nabla_i^2 + \sum_{i>j} \frac{e^2}{|r_i - r_j|} + \sum_{i} V_R(r_i)$$

The many-body Hamiltonian for the electrons is:

$$\hat{H} = -\frac{\hbar^2}{2m} \sum_{i} \nabla_i^2 + \sum_{i>j} \frac{e^2}{|r_i - r_j|} + \sum_{i} V_R(r_i)$$

The ground state energy (variational):

$$E[V] = \min_{\psi} \left(\langle \psi | \hat{K} + \hat{U} | \psi \rangle + \int \psi^* V_R(r) \psi dr \right) = \min_{\psi} \left(\langle \psi | \hat{K} + \hat{U} | \psi \rangle + \int \rho(r) V_R(r) dr \right)$$

The many-body Hamiltonian for the electrons is:

$$\hat{H} = -\frac{\hbar^2}{2m} \sum_{i} \nabla_i^2 + \sum_{i>j} \frac{e^2}{|r_i - r_j|} + \sum_{i} V_R(r_i)$$

The ground state energy (variational):

$$E[V] = \min_{\psi} \left(\langle \psi | \hat{K} + \hat{U} | \psi \rangle + \int \psi^* V_R(r) \psi dr \right) = \min_{\psi} \left(\langle \psi | \hat{K} + \hat{U} | \psi \rangle + \int \rho(r) V_R(r) dr \right)$$

• First Hohenberg-Kohn Theorem: $V \leftrightarrow \rho$

• The many-body Hamiltonian for the electrons is:

$$\hat{H} = -\frac{\hbar^2}{2m} \sum_{i} \nabla_i^2 + \sum_{i>j} \frac{e^2}{|r_i - r_j|} + \sum_{i} V_R(r_i)$$

• The ground state energy (variational):

$$E[V] = \min_{\psi} \left(\langle \psi | \hat{K} + \hat{U} | \psi \rangle + \int \psi^* V_R(r) \psi dr \right) = \min_{\psi} \left(\langle \psi | \hat{K} + \hat{U} | \psi \rangle + \int \rho(r) V_R(r) dr \right)$$

- First Hohenberg-Kohn Theorem: $V \leftrightarrow \rho$
- Second Hohenberg-Kohn Theorem: $E[V] = \min_{\rho} \left(F[\rho] + \int_{\rho} \rho(r) V(r) dr \right)$

Kohn-Sham potential

• Take the functional derivative of $E[\rho]$ with respect to ρ :

$$F[\rho] = K[\rho] + \frac{1}{2} \int \frac{\rho(r)\rho(r')}{|r - r'|} dr dr' + E_{xc}[\rho]$$

Kohn-Sham potential

• Take the functional derivative of $E[\rho]$ with respect to ρ :

$$F[\rho] = K[\rho] + \frac{1}{2} \int \frac{\rho(r)\rho(r')}{|r - r'|} dr dr' + E_{xc}[\rho]$$

$$\frac{\delta E[\rho]}{\delta \rho(r)} = \frac{\delta K[\rho]}{\delta \rho(r)} + \int \frac{\rho(r')}{|r - r'|} dr' + \frac{\delta E_{\chi c}}{\delta \rho} + V(r)$$

Kohn-Sham potential

• Take the functional derivative of E[
ho] with respect to ho :

$$F[\rho] = K[\rho] + \frac{1}{2} \int \frac{\rho(r)\rho(r')}{|r - r'|} dr dr' + E_{xc}[\rho]$$

$$\frac{\delta E[\rho]}{\delta \rho(r)} = \frac{\delta K[\rho]}{\delta \rho(r)} + \int \frac{\rho(r')}{|r - r'|} dr' + \frac{\delta E_{xc}}{\delta \rho} + V(r)$$

 ${
m V}_{KS}[
ho]$ Kohn-Sham potential

What is in E_x ?

Local Density Approximation (LDA):

$$E_{xc} = \int \epsilon_{xc}(\rho(r))\rho(r)dr$$

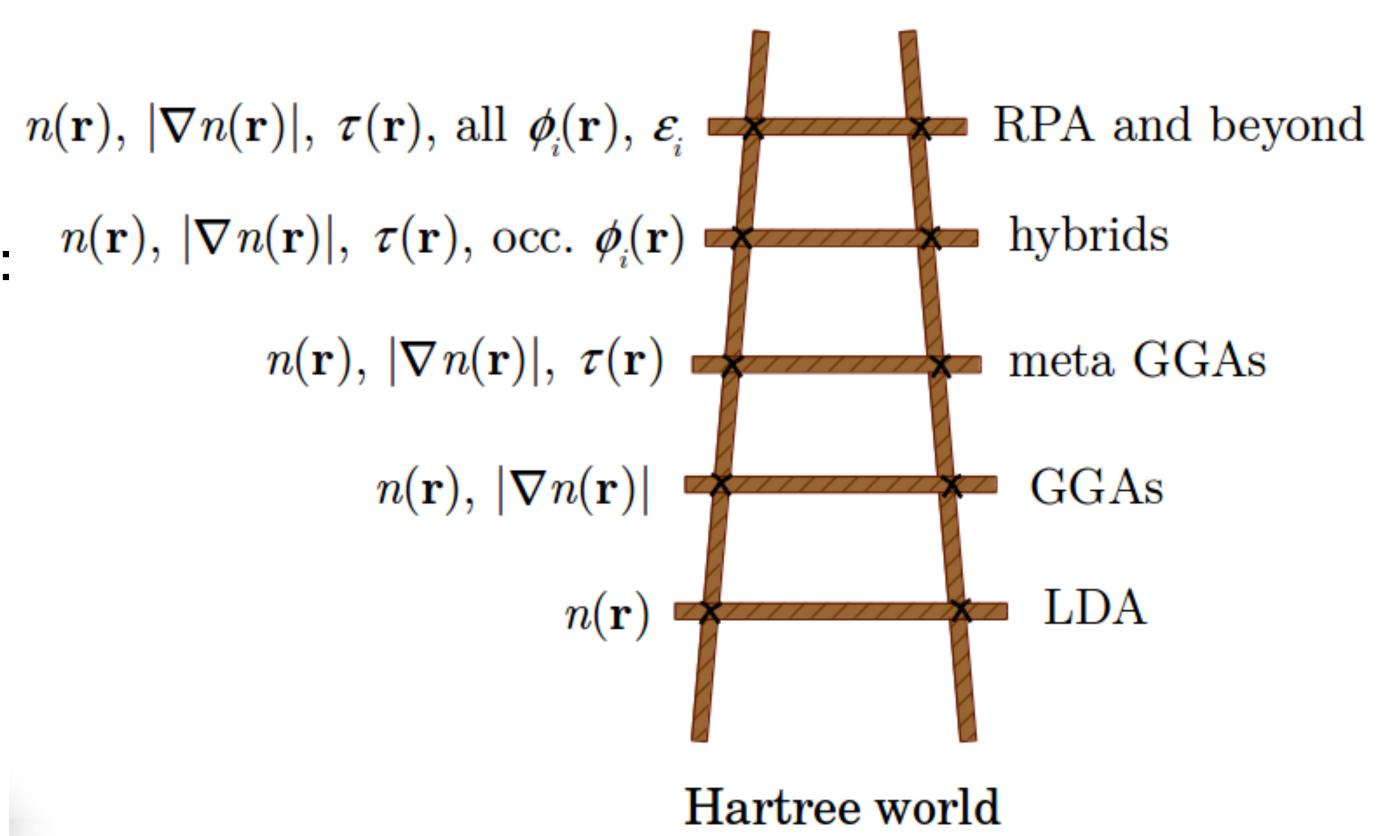
• Generalized Gradient Approx. (GGA):
$$E_{xc} = \int \epsilon_{xc}(\rho(r), \nabla(\rho))\rho(r)dr$$

vdW functionals:

$$E_{vdW} = \int \rho(r)\rho(r')\Phi_{vdW}(r,r')drdr'$$

Hybrid functionals

Heaven of chemical accuracy



Kohn-Sham equations

$$\frac{\delta E[\rho]}{\delta \rho(r)} = \frac{\delta K[\rho]}{\delta \rho(r)} + \int \frac{\rho(r')}{|r - r'|} dr' + \frac{\delta E_{xc}}{\delta \rho} + V(r)$$

$${
m V}_{KS}[
ho]$$
 Kohn-Sham potential

$$\rho(r) = \sum_{v} |\psi_{v}(r)|^{2}$$

Represent the density in terms of singleparticle wavefunctions.

The sum runs over the occupied states.

$$\left(-\frac{\hbar^2}{2m}\nabla^2 + \mathbf{v}_{KS}[\rho]\right)\psi_v(r) = \epsilon_v\psi_v(r)$$
 Self-consistent problem.

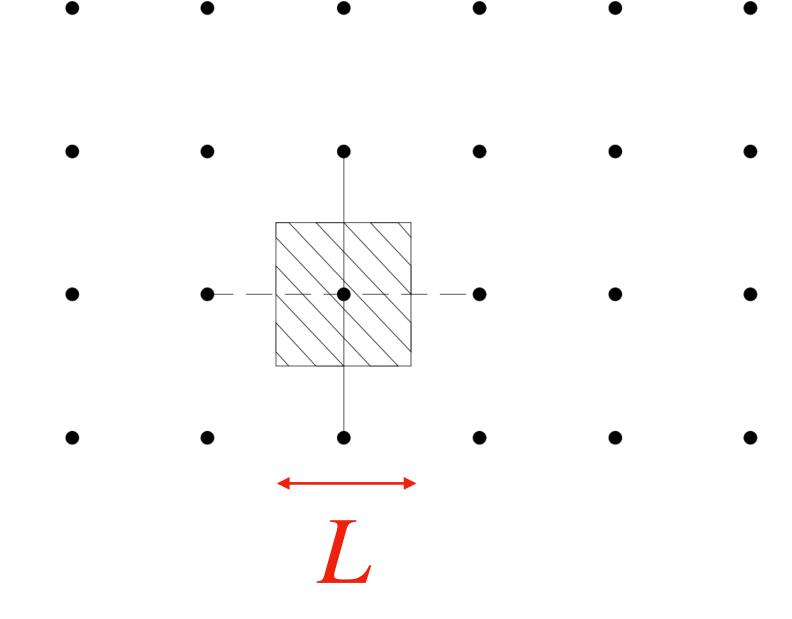
To derive the KS equations one can apply the variational principle with respect to the noninteracting wavefunctions (Kohn-Sham): $\psi_{v}(r)$

• We need to express $\psi(r)$ on basis set:

$$\psi_{\mathbf{v}}(r) = \sum_{j} c_{j,\mathbf{v}} \phi_{j}(r)$$

• We need to express $\psi(r)$ on basis set:

$$\psi_{\mathbf{v}}(r) = \sum_{j} c_{j,\mathbf{v}} \phi_{j}(r)$$

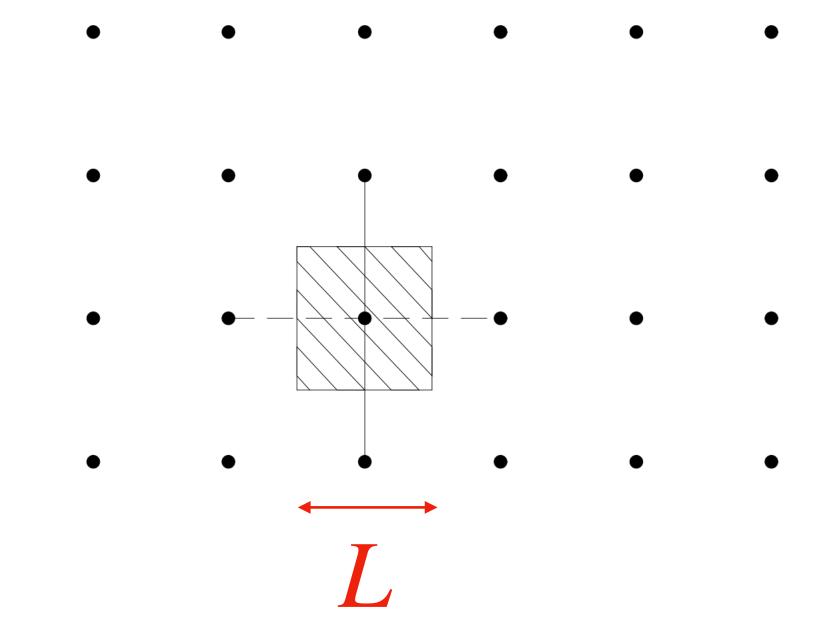


• We need to express $\psi(r)$ on basis set:

$$\psi_{\mathbf{v}}(r) = \sum_{j} c_{j,\mathbf{v}} \phi_{j}(r)$$

Plane waves naturally comply with crystal periodicity

$$\phi_j = e^{iq_j \cdot r}; \quad q_j = \frac{2\pi}{L} j \text{ with } j \in \mathbb{Z}$$

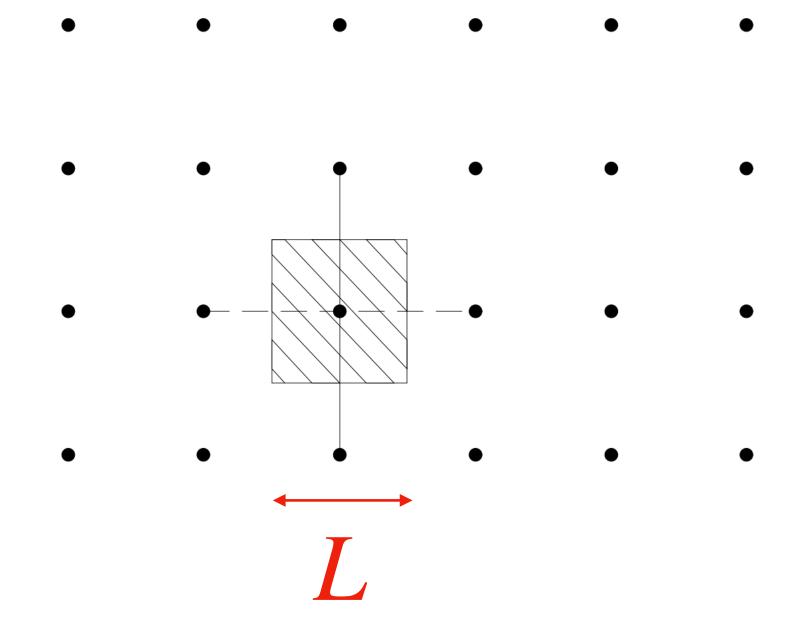


• We need to express $\psi(r)$ on basis set:

$$\psi_{\mathbf{v}}(r) = \sum_{j} c_{j,\mathbf{v}} \phi_{j}(r)$$

Plane waves naturally comply with crystal periodicity

$$\phi_j = e^{iq_j \cdot r}; \quad q_j = \frac{2\pi}{L} j \text{ with } j \in \mathbb{Z}$$



How many PW shall we take?

$$\frac{\hbar^2}{2m}q_j^2 \le E_{cut}$$

Brillouin Zone Sampling: k-points

Bloch Theorem

$$\psi_k(x+L) = e^{ikL}\psi_k(x)$$

$$\psi_k(x) = e^{ikx} u_k(x)$$

$$u_k(x+L) = u_k(x)$$

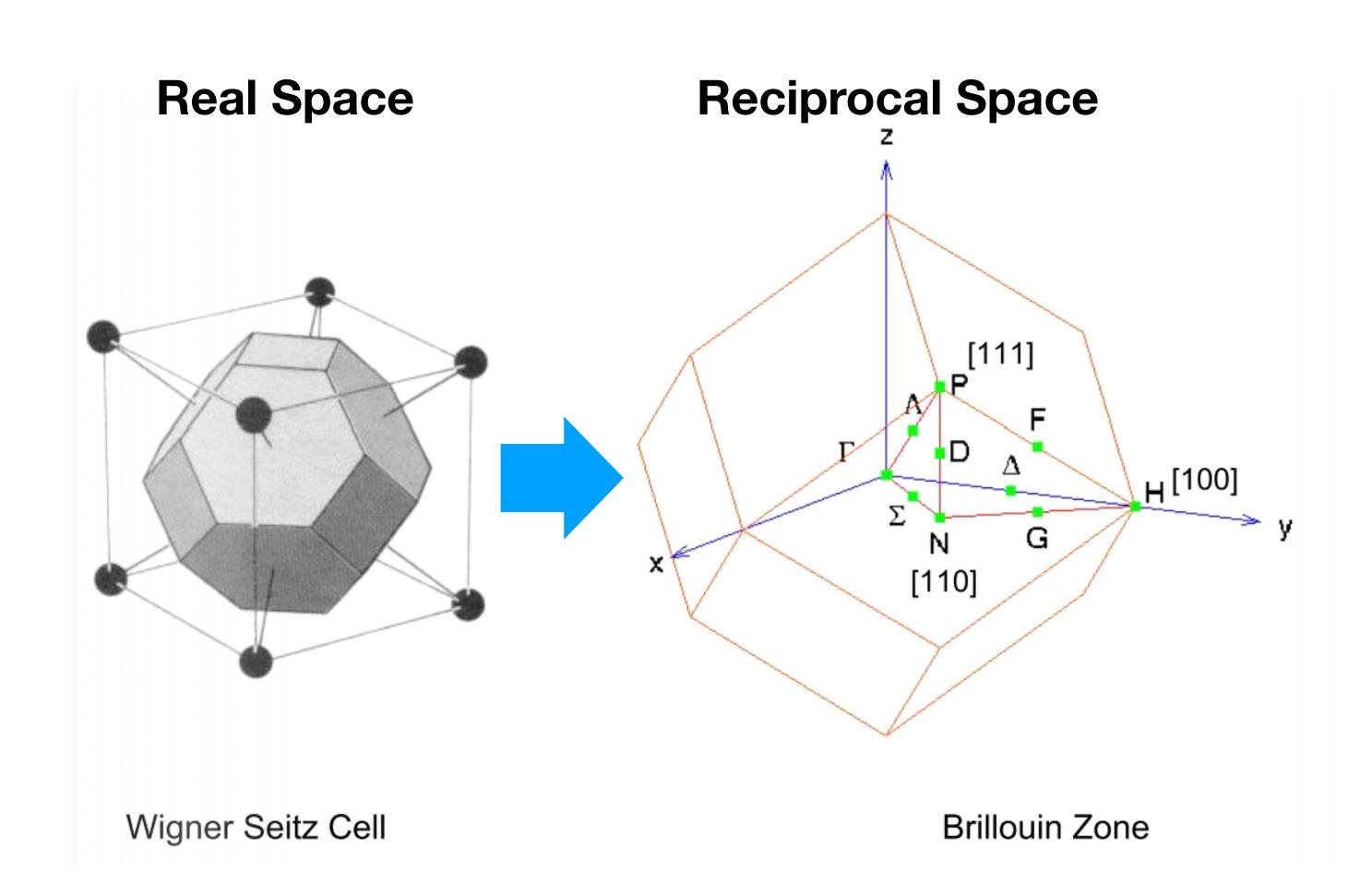
Brillouin Zone Sampling: k-points

Bloch Theorem

$$\psi_k(x+L) = e^{ikL}\psi_k(x)$$

$$\psi_k(x) = e^{ikx} u_k(x)$$

$$u_k(x+L) = u_k(x)$$



Brillouin Zone Sampling: k-points

Bloch Theorem

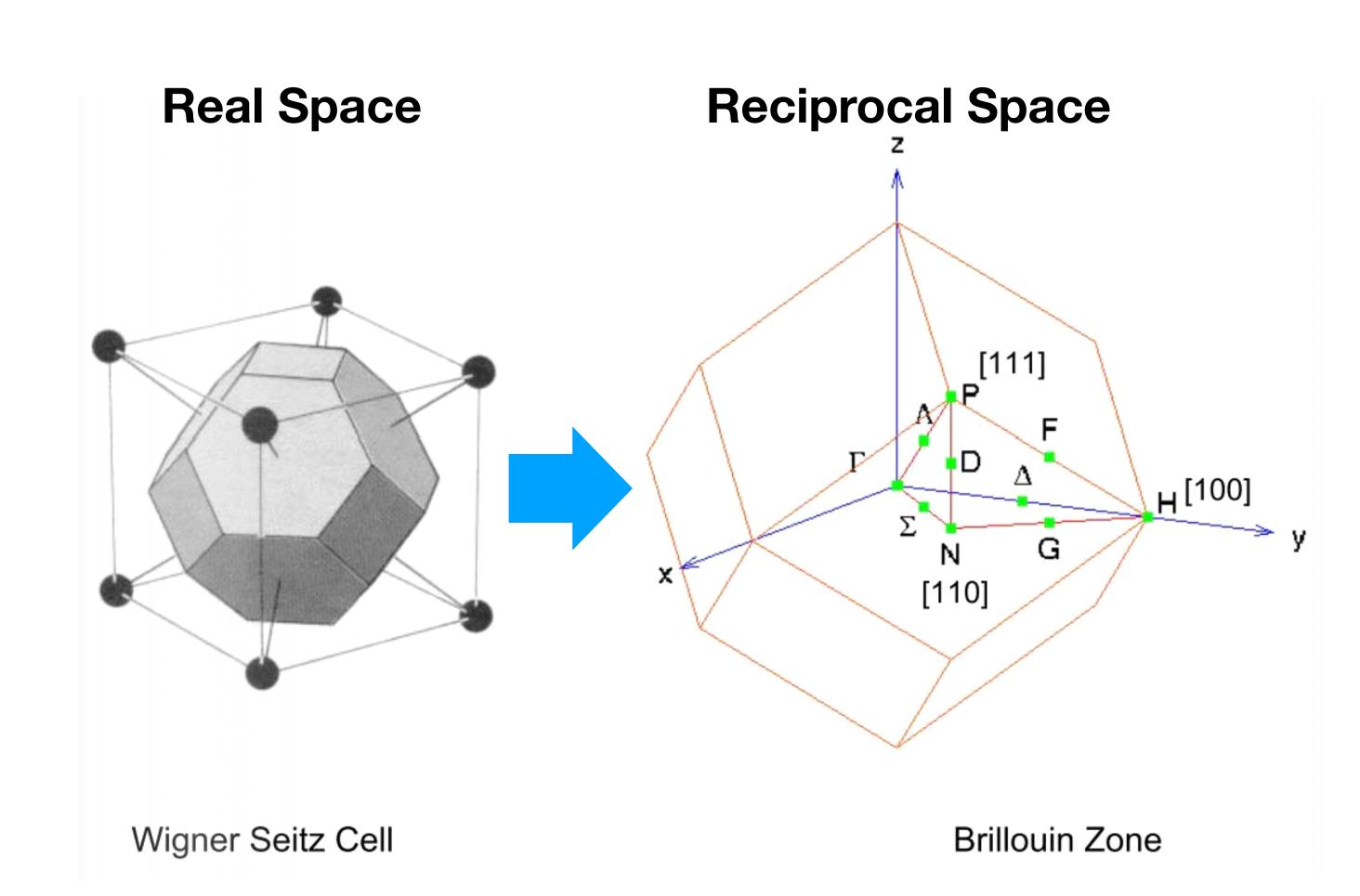
$$\psi_k(x+L) = e^{ikL}\psi_k(x)$$

$$\psi_k(x) = e^{ikx} u_k(x)$$

$$u_k(x+L) = u_k(x)$$

Using plane waves:

$$\psi_{v,k}(r) = e^{ik \cdot r} \sum_{G} c_{v,k}(G) e^{iG \cdot r}$$



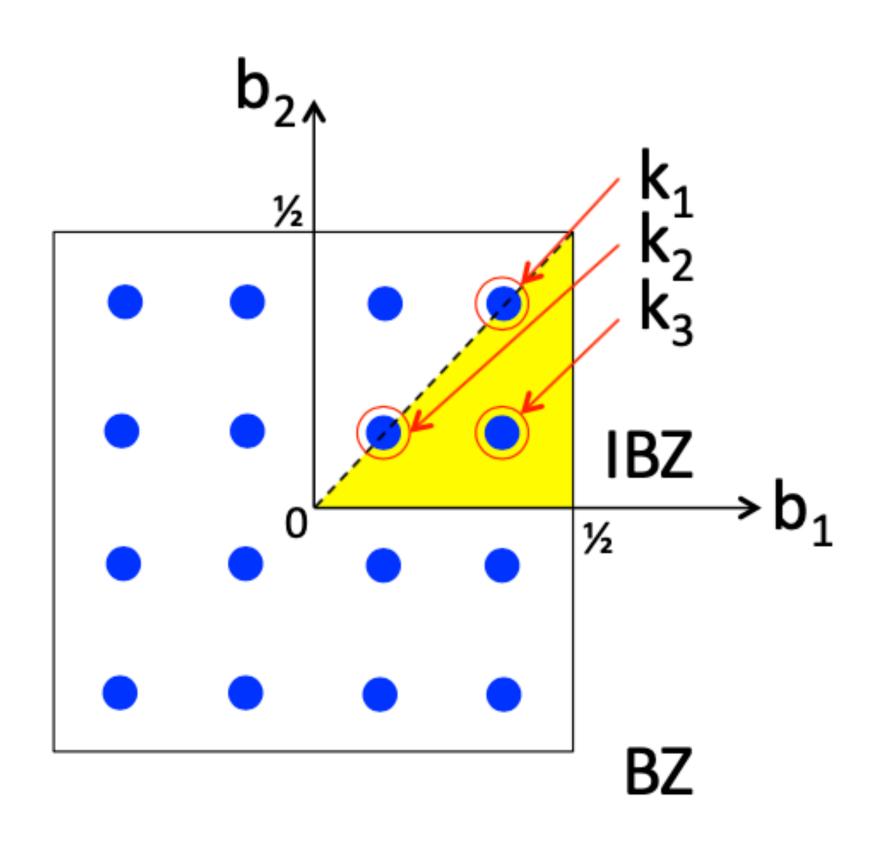
Efficient BZ sampling

Observables are averaged over k points

$$\langle A \rangle = \frac{1}{\Omega_{BZ}} \int_{BZ} dk A(k) = \frac{1}{N_k} \sum_{i}^{N_k} A(k_i)$$

For example, th total energy is:

$$E_{tot} = \sum_{i}^{N_k} E_i w_i$$



It is often conveniente to "shift" the k-point mesh

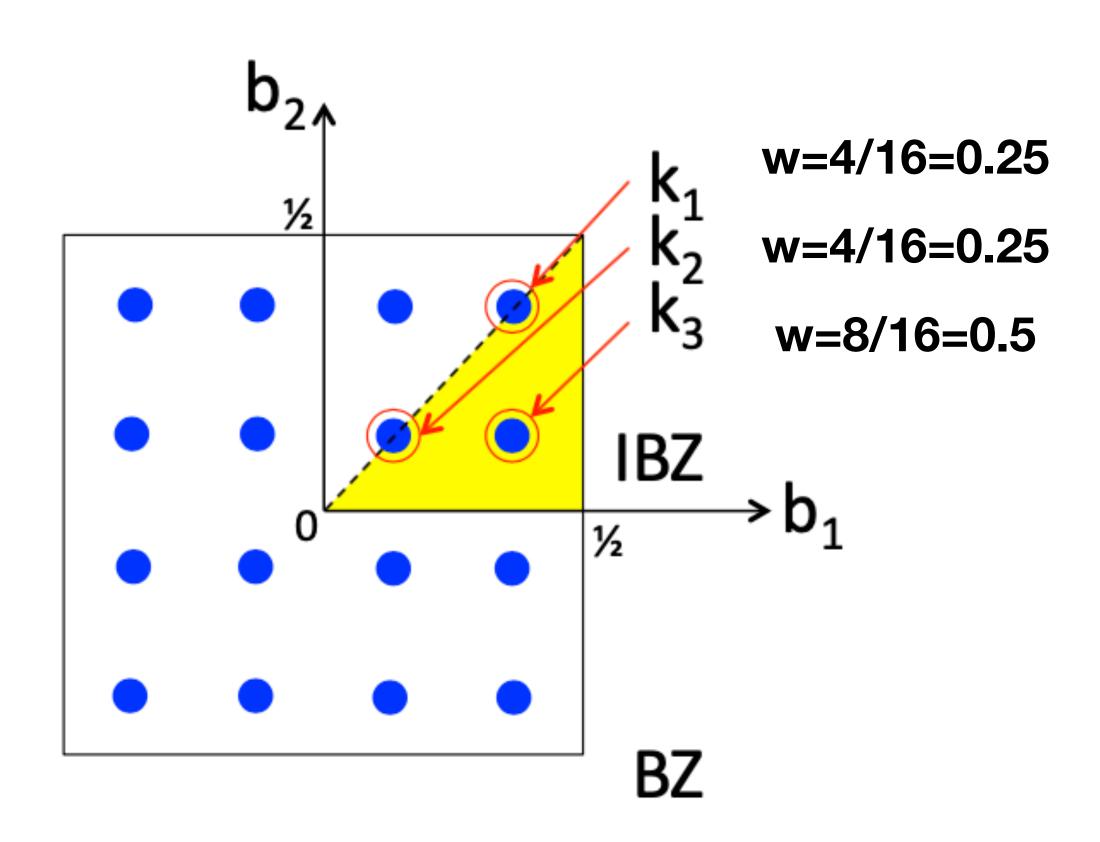
Efficient BZ sampling

Observables are averaged over k points

$$\langle A \rangle = \frac{1}{\Omega_{BZ}} \int_{BZ} dk A(k) = \frac{1}{N_k} \sum_{i}^{N_k} A(k_i)$$

For example, th total energy is:

$$E_{tot} = \sum_{i}^{N_k} E_i w_i$$



It is often conveniente to "shift" the k-point mesh

Advantages of using plane waves

- ullet Convergence checked by tuning only E_{cut}
- Basis set independent of nuclear positions: no "Pulay forces"
- Use Fourier transform to solve Poisson equation: $\nabla^2 V = 4\pi \rho$
- Density, Hartree, and XC potentials easily calculated
- PWs are orthonormal (normalization: $1/\sqrt{\Omega}$)

Disadvantages of using plane waves

 The basis set depends on the shape/size of the cell: Pulay Stress

 One would need a huge number of PW to treat localized (core) electrons and nodes of valence electrons

Disadvantages of using plane waves

 The basis set depends on the shape/size of the cell: Pulay Stress

Fix: See hands on tutorial

 One would need a huge number of PW to treat localized (core) electrons and nodes of valence electrons

Disadvantages of using plane waves

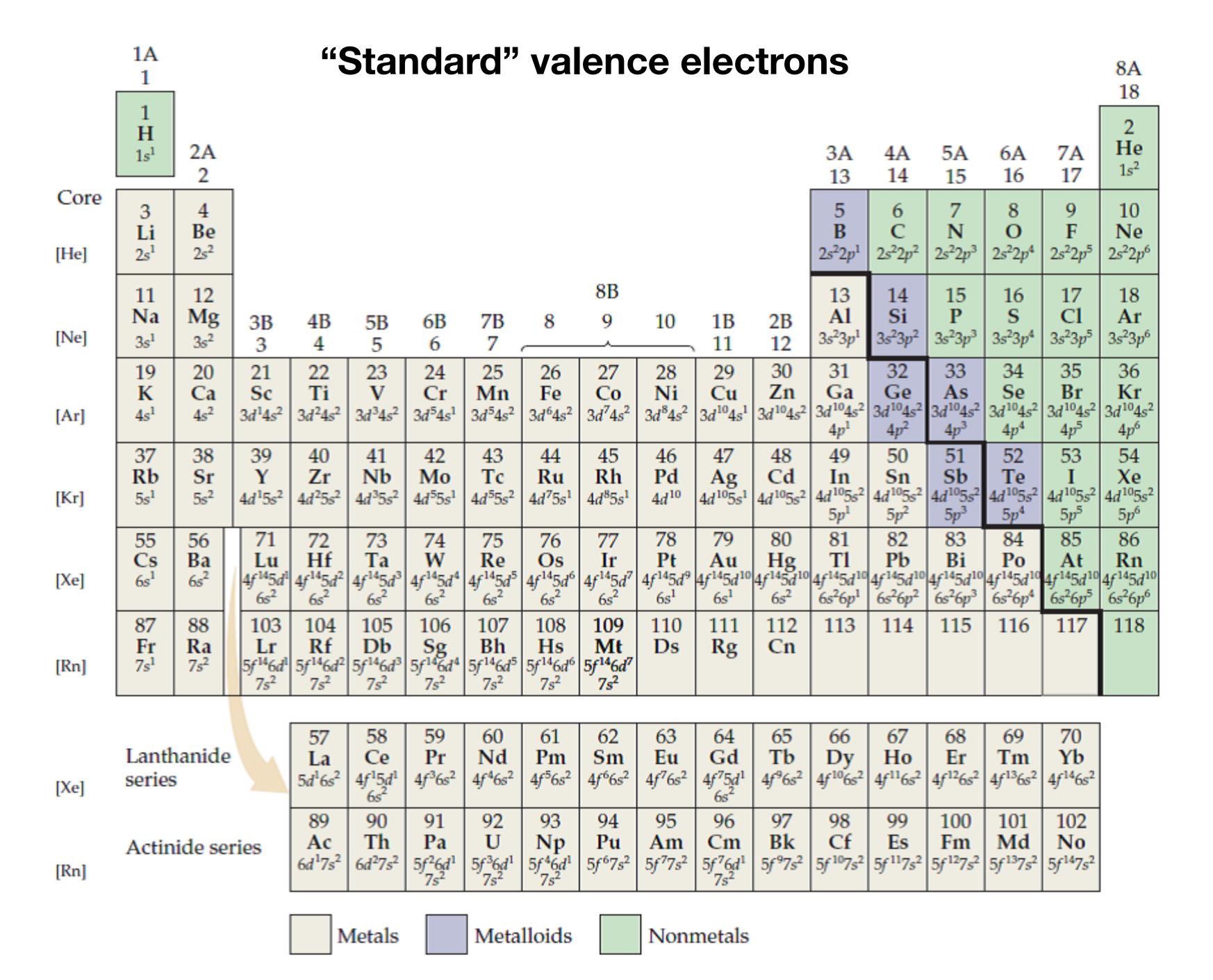
 The basis set depends on the shape/size of the cell: Pulay Stress

Fix: See hands on tutorial

 One would need a huge number of PW to treat localized (core) electrons and nodes of valence electrons

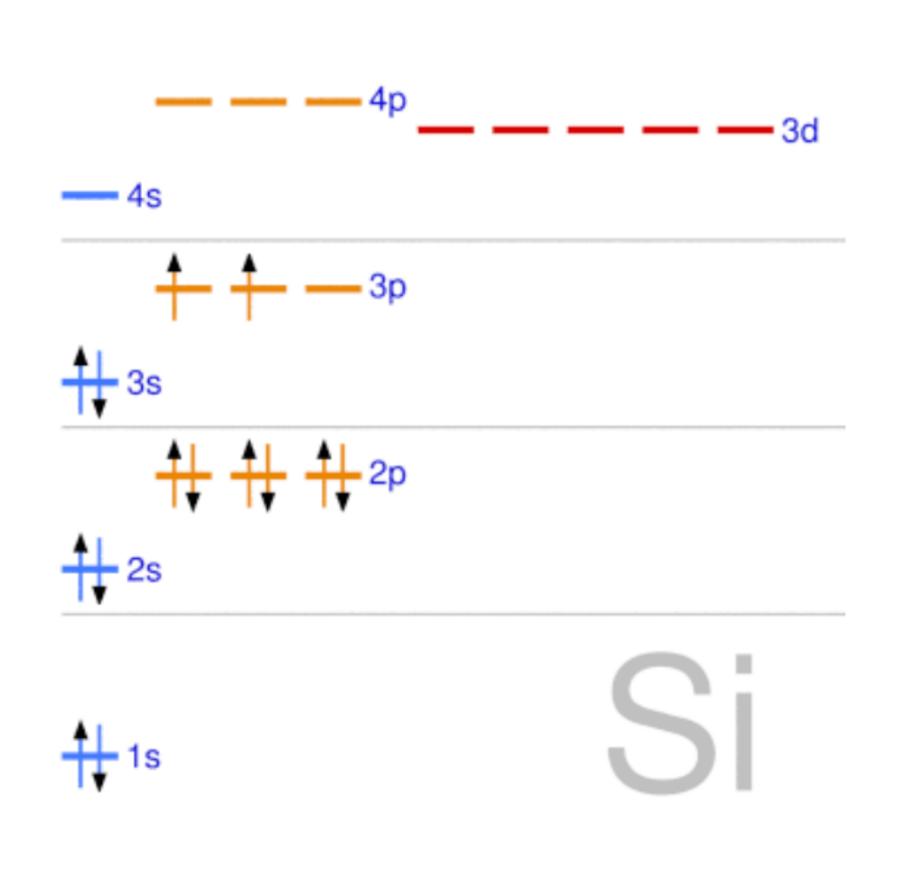
Fix: Pseudopotentials

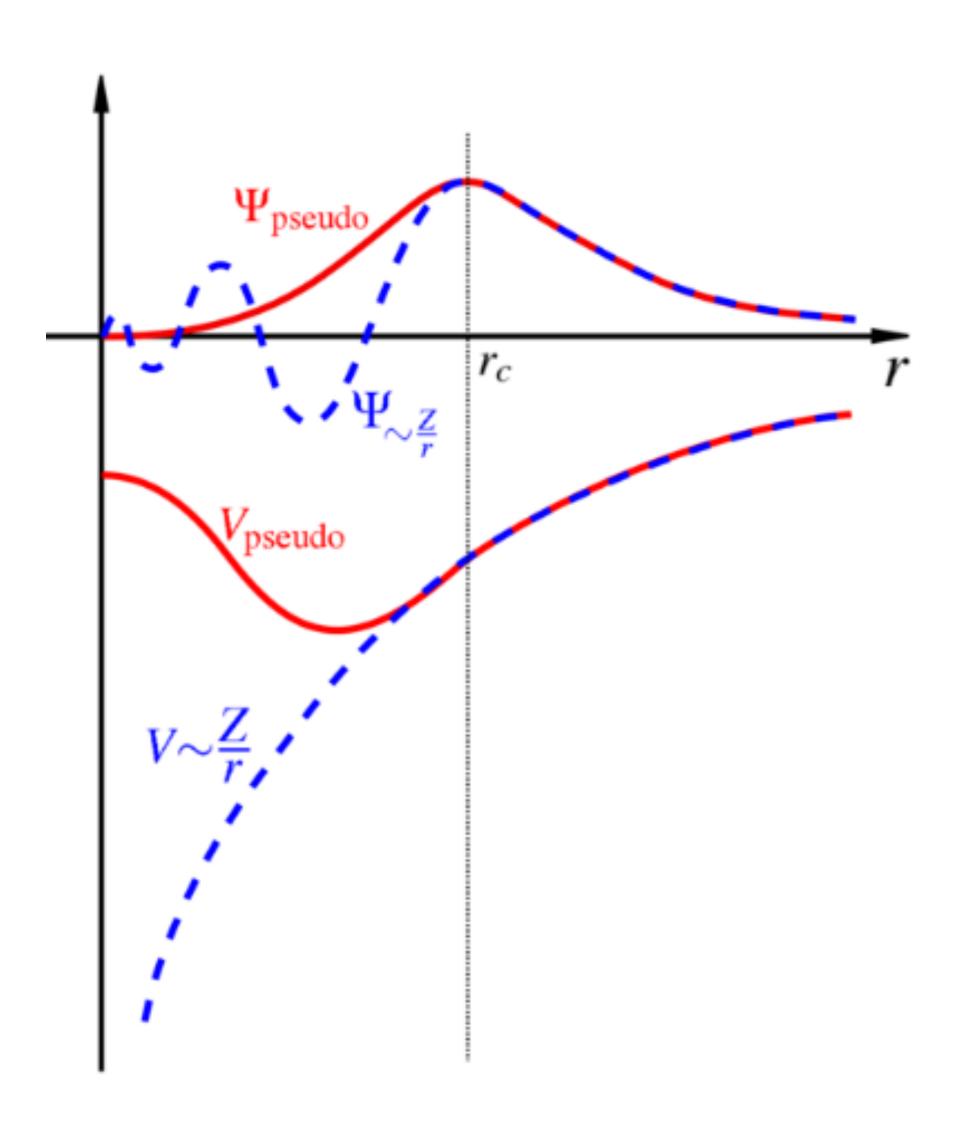
Dumping core states

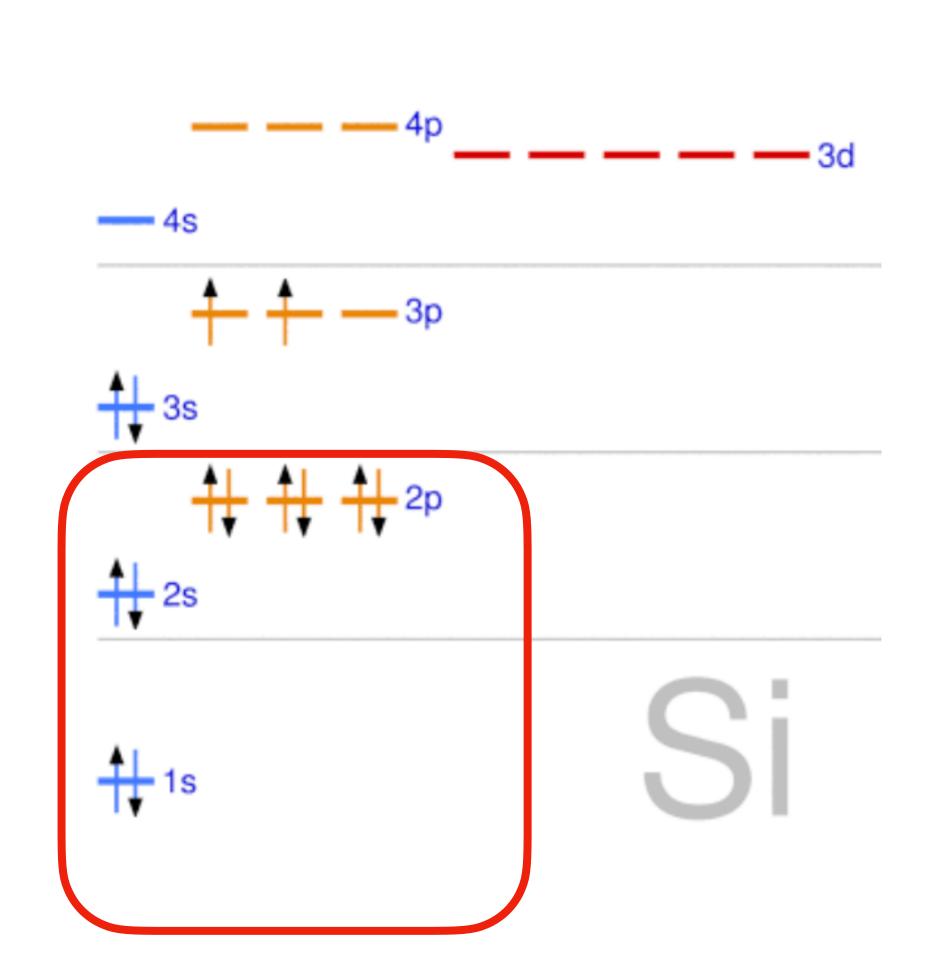


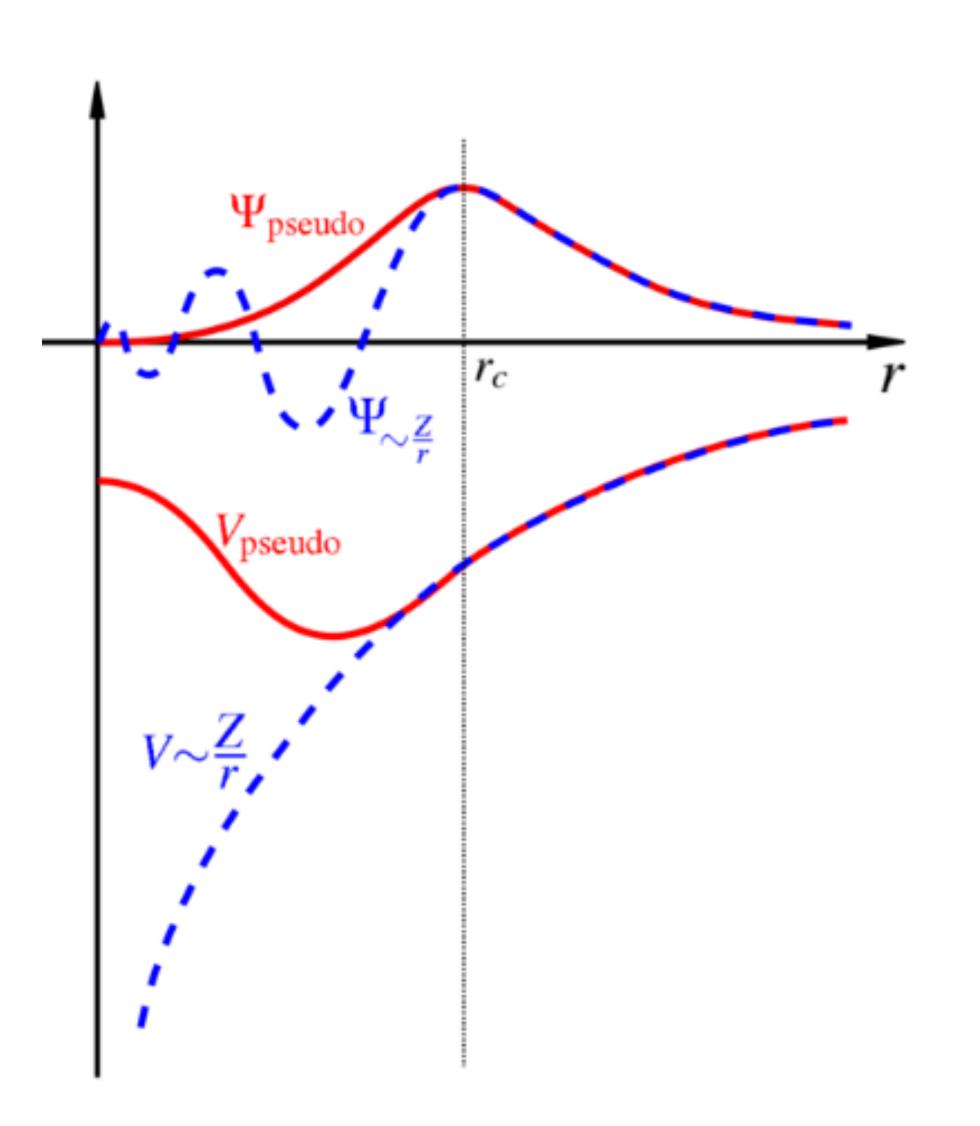
Dumping core states

- Define a "pseudo-atom" = nucleus + core states (screened nucleus)
- Valence states are the lowest states with a given angular symmetry:
 - ψ^{pp} is smooth and has no nodes
- The chemical behavior of the pseudo-atom has to be the same as the actual atom:
 - Same orbital energies: $\epsilon_{\rm v}^{PP} = \epsilon_{\rm v}^{AE}$
 - Same wave function outside the "nuclear region": $\psi_{\rm v}^{pp}(r) = \psi_{\rm v}^{AE}(r)$ for $r > r_C$

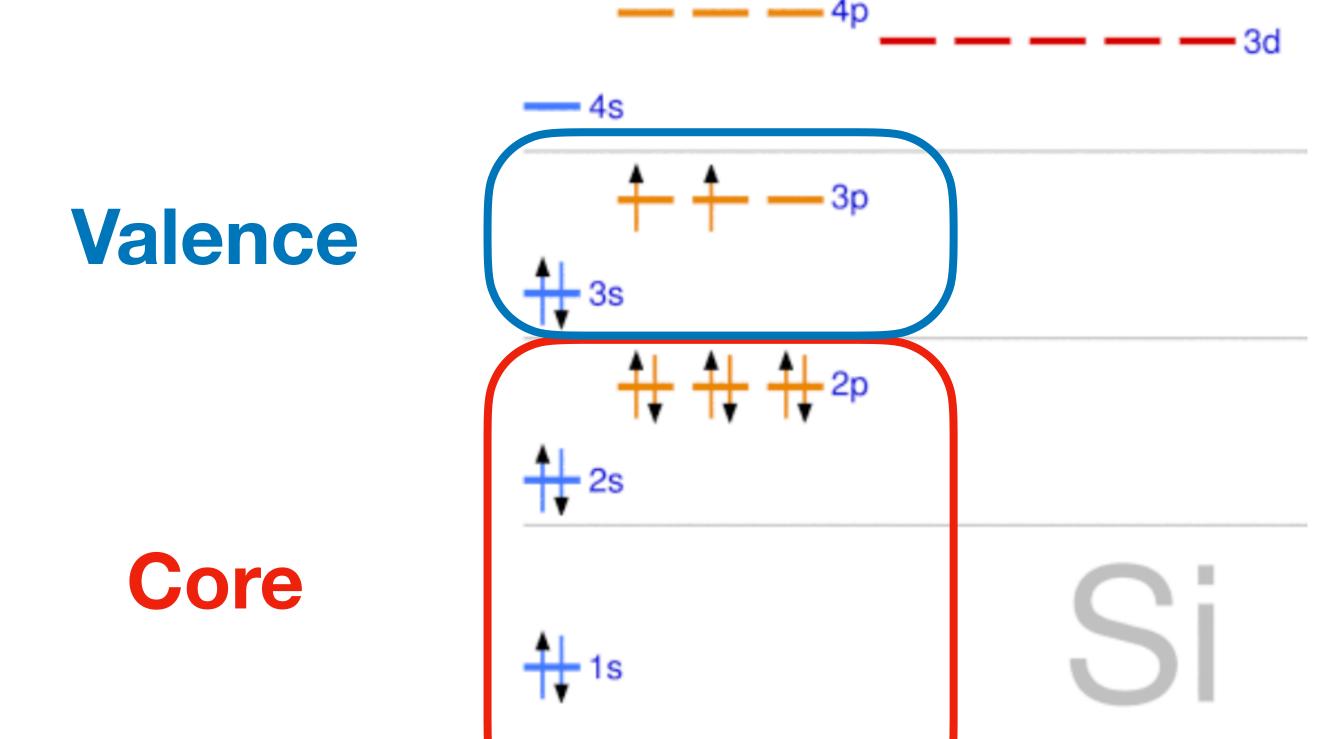


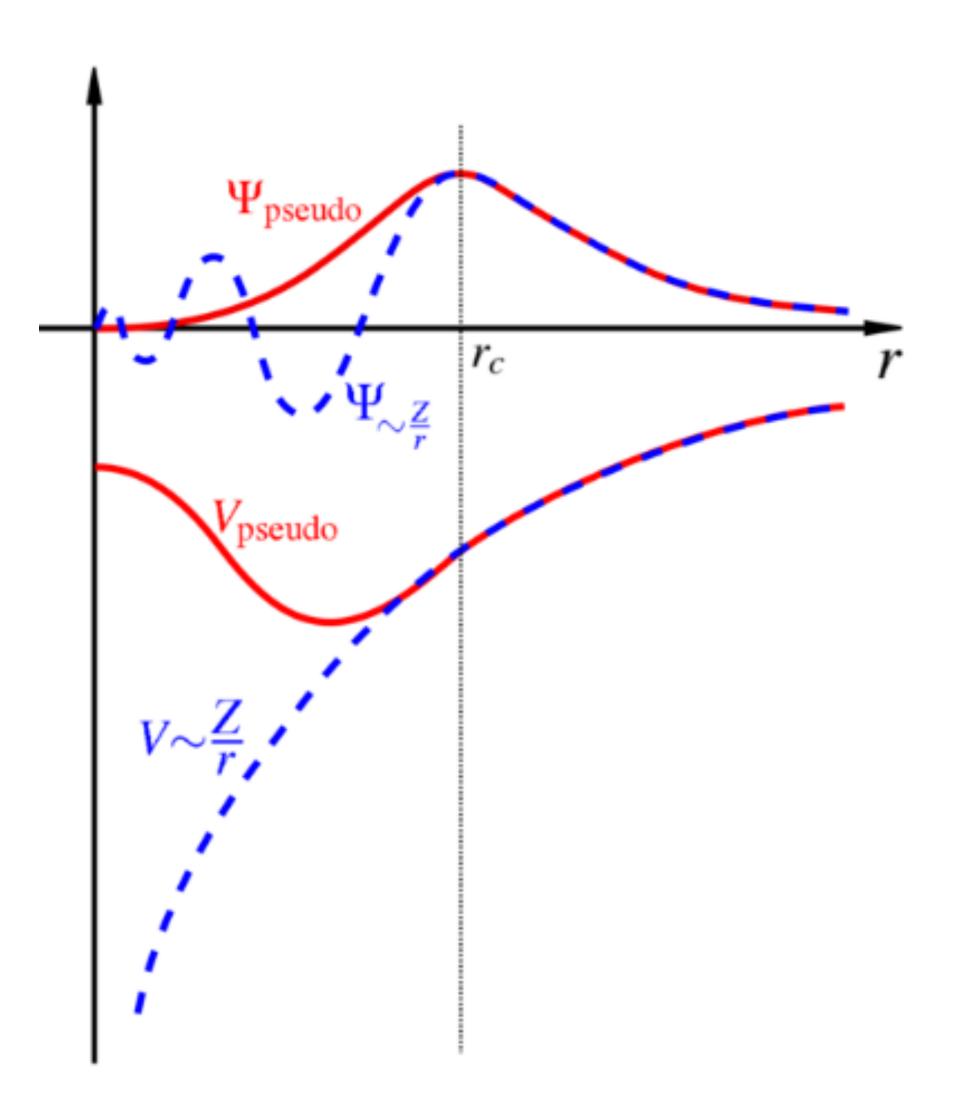




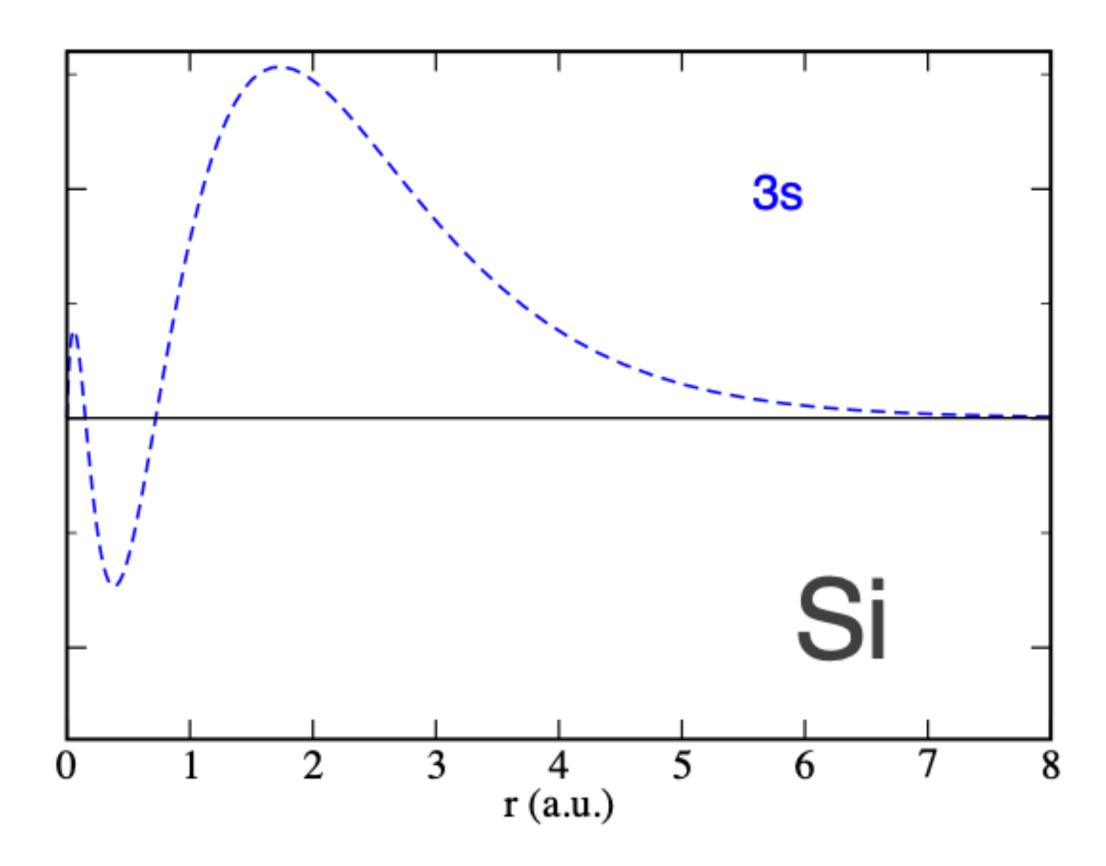


Core



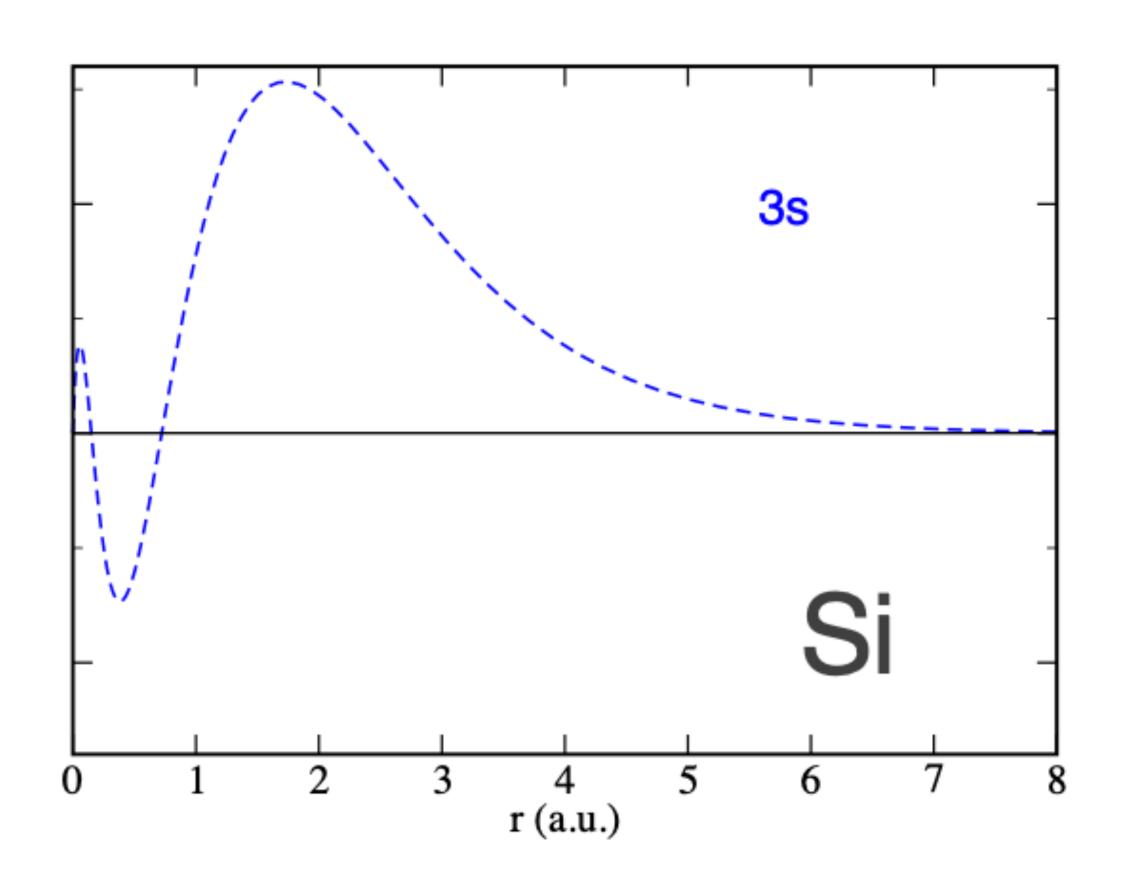


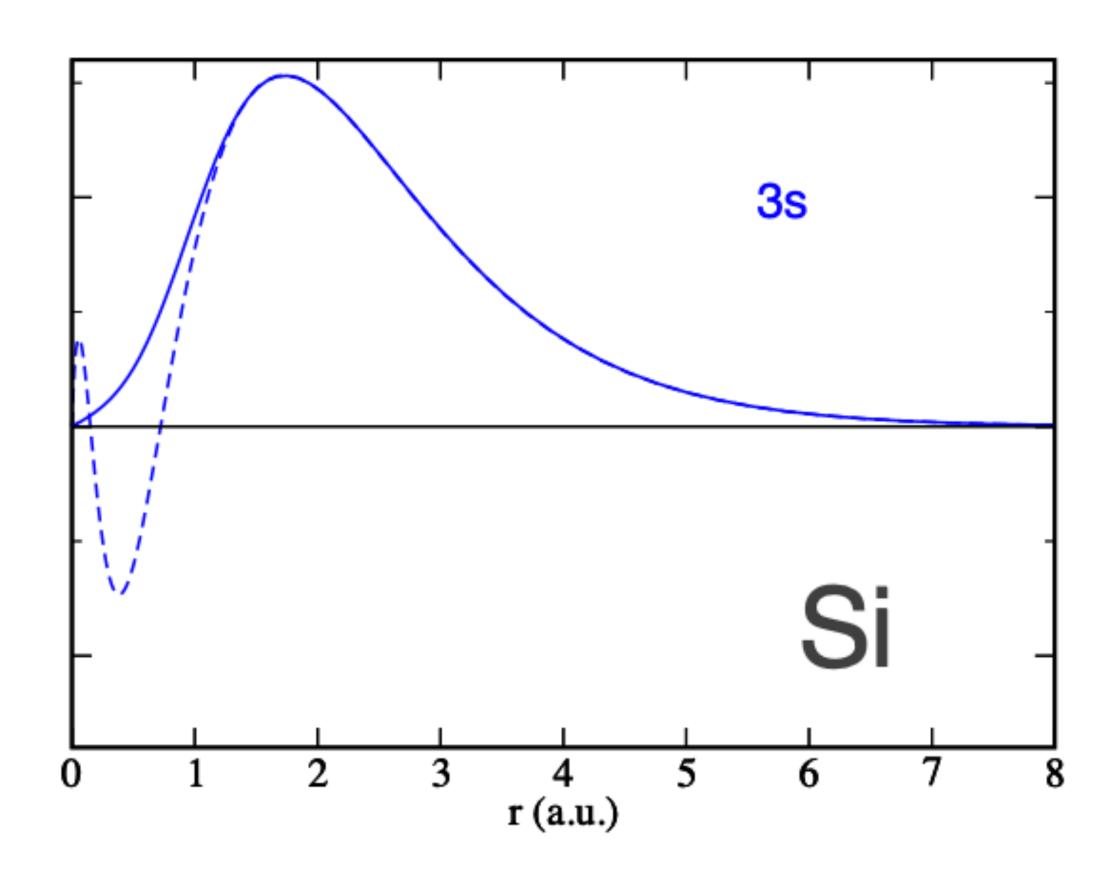
Real 3s WF



Real 3s WF



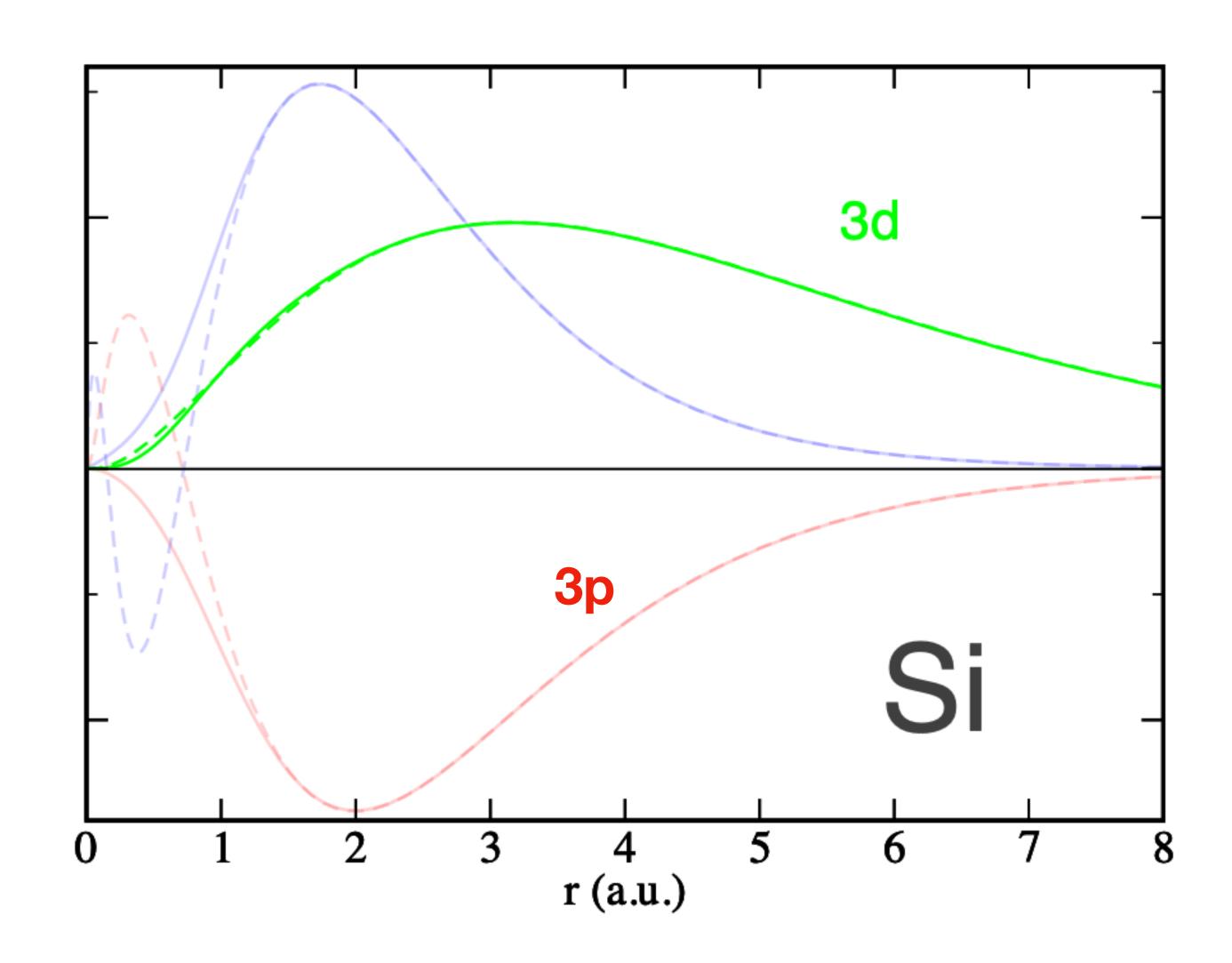




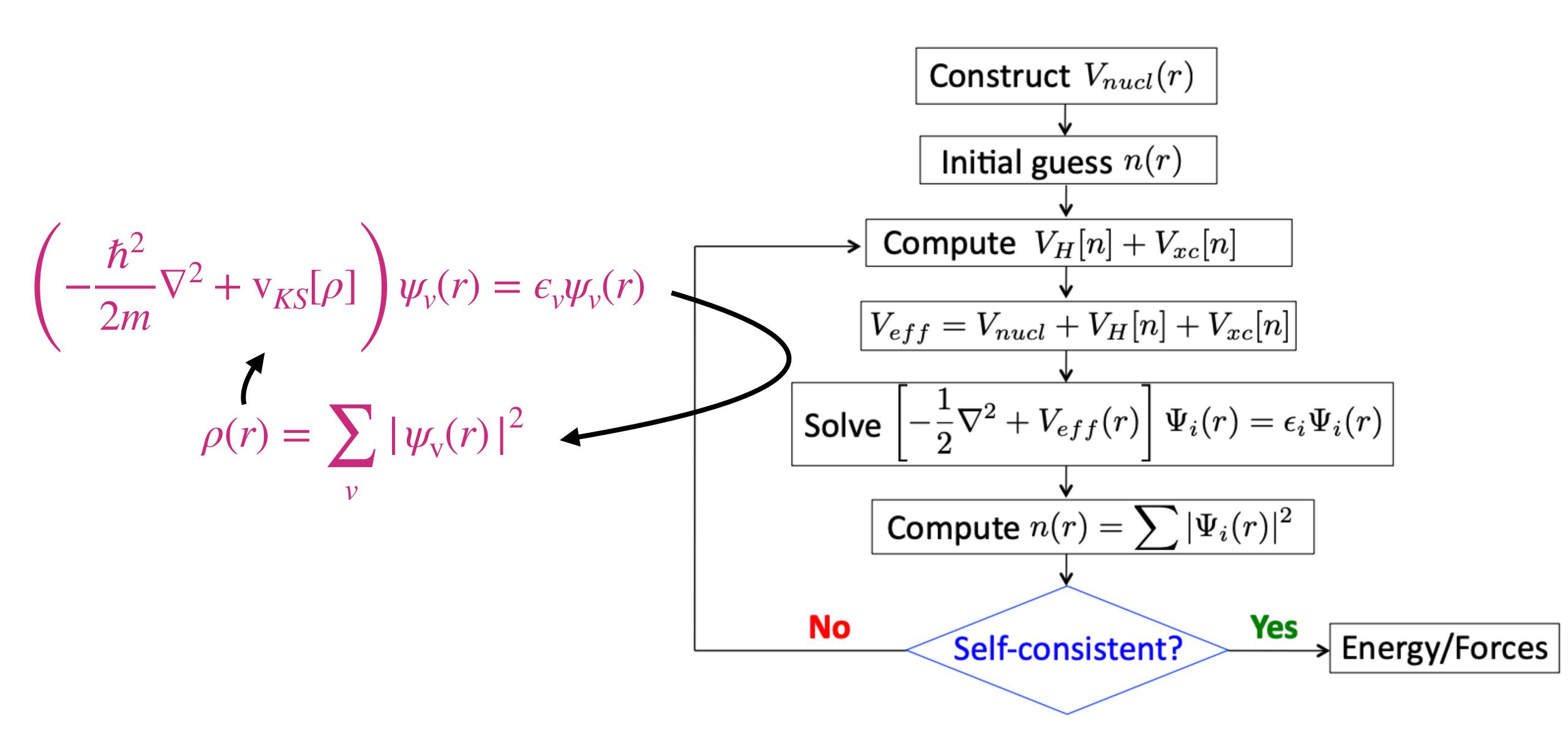
Pseudo wavefunctions of Si

3p and 3d* channels

* 3d is empty



Self consistent calculation



Hands-on Tutorial

Where to start

https://www.quantum-espresso.org/pseudopotentials

PSEUDOPOTENTIALS

More about pseudopotentials

SSSP on Materials Cloud

Pseudo DoJo

ONCV Potentials

SCAN pseudopotentials

PSlibrary table

Original QE PP table

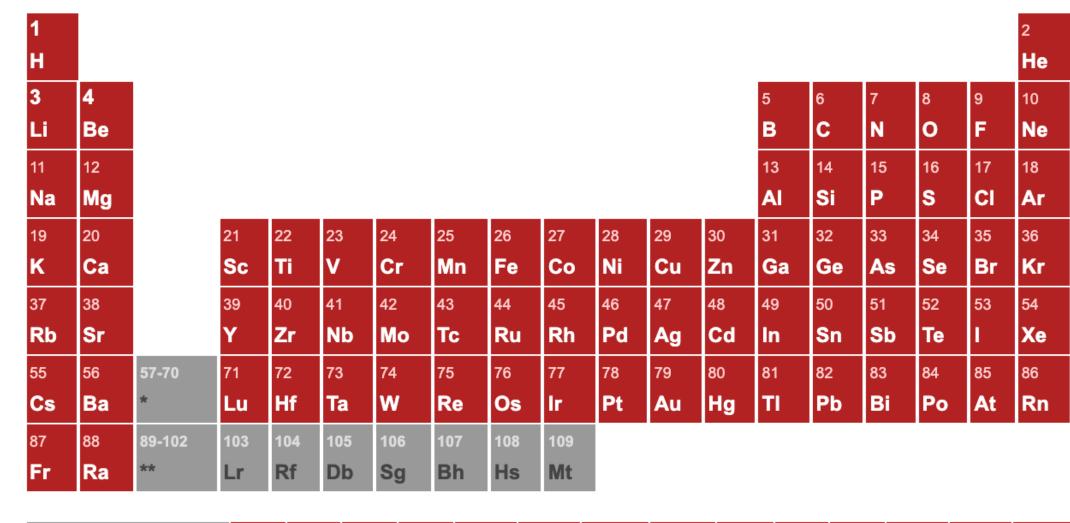
Hartwigesen-Goedecker-Hutter PP table

Old FHI PP table

PSLIBRARY

Ready-to-use pseudopotentials from PSlibrary (recommended). For other ready-to-use tables, follow the links of the menu at the left. For more info, see here.

Please cite the pseudopotentials used and give proper credit to their authors (see this page for a rather complete list of acknowledgments).



*	57	58	59	60	61	62	63	64	65	66	67	68	69	70
Lanthanoids	La	Се	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Но	Er	Tm	Yb
**	89	90	91	92	93	94	95	96	97	98	99	100	101	102
Actinoids	Ac	Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No

Get pseudopotentials:

Pseudopotentials file

```
<UPF version="2.0.1">
  <PP INFO>
Generated using "atomic" code by A. Dal Corso v.6.3
Author: ADC
Generation date: 6Sep2018
Pseudopotential type: PAW
Element: Ga
Functional: SLA PW
                      PBX PBC
    Suggested minimum cutoff for wavefunctions: 43. Ry
    Suggested minimum cutoff for charge density: 240. Ry
    The Pseudo was generated with a Scalar-Relativistic Calculation
    Local Potential by smoothing AE potential with Bessel fncs, cutoff radius:
                                                                                 1.9000
    Pseudopotential contains additional information for GIPAW reconstruction.
    Valence configuration:
   nl pn l
              occ
                                 Rcut US
                                               E pseu
                         Rcut
                                            -0.657693
                                   2.200
             2.00
                       1.300
       2 1 1.00
                       1.300
                                   2.200
                                            -0.189560
    3D 3 2 10.00
                       1.300
                                  1.700
                                            -1.407827
    Generation configuration:
                                   2.200
                                            -0.657691
             2.00
                       1.300
                                   2.200
                       1.300
                                             6.100000
             0.00
                                   2.200
       2 1 1.00
                       1.300
                                            -0.189560
       2 1 0.00
                       1.300
                                   2.200
                                             6.300000
                                  1.700
       3 2 10.00
                       1.300
                                            -1.407824
    3D
          2 0.00
                       1.300
                                  1.700
                                            -1.300000
    3D
    Pseudization used: troullier-martins
```

```
<PP INPUTFILE>
 & input
  title='Ga',
  zed=31.,
  rel=1,
  config='[Ar] 4s2 4p1 3d10',
  iswitch=3,
  dft='PBE'
 & inputp
  lpaw = .true.,
  use xsd=.FALSE.,
   pseudotype=3,
  file pseudopw='Ga.pbe-dnl-kjpaw psl.1.0.0.UPF',
   author='ADC',
  11oc=-1,
  rcloc=1.9,
  which augfun='PSQ',
  rmatch_augfun_nc=.true.,
  nlcc=.true.,
  new_core_ps=.true.,
  rcore=1.1,
  tm=.true.
         2.00 0.00 1.30 2.20 0.0
        0.00 6.10 1.30 2.20 0.0
   2 1 1.00 0.00 1.30 2.20 0.0
4P 2 1 0.00 6.30 1.30 2.20 0.0
3D 3 2 10.00 0.00 1.30 1.70 0.0
3D 3 2 0.00 -1.30 1.30 1.70 0.0
</PP INPUTFILE>
 </PP_INFO>
 <!--
END OF HUMAN READABLE SECTION
```

Input file example: GaAs

Namelist (FORTRAN)

&electrons

Sections

Variables

```
&control
    calculation = 'scf'
    restart_mode='from_scratch',
    prefix='GaSb',
    tstress = .true.
    tprnfor = .true.
    pseudo_dir = '../../Pseudo/',
    outdir='./tmp'
/
&system
    ibrav= 2,
    celldm(1) = 10.7,
```

ecutwfc = 43.0, ! energy in Ry

ecutrho = 300. ! energy in Ry

nat = 2,

ntyp=2,

```
diagonalization='david'
  conv_{thr} = 1.0e-8
  mixing_beta = 0.4
ATOMIC_SPECIES
 Ga 69.723000d0 Ga.pbe-dnl-kjpaw_psl.1.0.0.UPF
 As 74.921600d0 As.pbe-n-kjpaw_psl.1.0.0.UPF
ATOMIC_POSITIONS crystal
Ga -0.125 -0.125 -0.125
As 0.125 0.125 0.125
K_POINTS automatic
222111
```

Running pw.x

mpirun -np 4 pw.x -nk 2 < scf.in > scf.out

For bigger systems (and bigger computers)

mpirun -np 4096 pw.x -nk 8 -nt 4 -nd 144 < <u>scf.in</u>

k-points
Tasks for 3D FFT

Subspace diagonalization

```
Program PWSCF v.6.4.1 starts on 60ct2021 at 18: 1:36
This program is part of the open-source Quantum ESPRESSO suite
for quantum simulation of materials; please cite
    "P. Giannozzi et al., J. Phys.:Condens. Matter 21 395502 (2009);
    "P. Giannozzi et al., J. Phys.:Condens. Matter 29 465901 (2017);
    URL http://www.quantum-espresso.org",
in publications or presentations arising from this work. More details at
http://www.quantum-espresso.org/quote
Parallel version (MPI), running on
                                       1 processors
MPI processes distributed on
                                 1 nodes
Waiting for input...
Reading input from standard input
Current dimensions of program PWSCF are:
Max number of different atomic species (ntypx) = 10
Max number of k-points (npk) = 40000
Max angular momentum in pseudopotentials (lmaxx) = 3
          file Ga.pbe-dnl-kjpaw psl.1.0.0.UPF: wavefunction(s) 4S 3D renormalized
Subspace diagonalization in iterative solution of the eigenvalue problem:
a serial algorithm will be used
G-vector sticks info
                                                                   PW
         dense
                                   G-vecs:
                                                       smooth
           1177
                           199
                    673
                                              26813
                                                       11575
                                                                 1807
bravais-lattice index
lattice parameter (alat) =
                                 10.7000 a.u.
unit-cell volume
                                306.2607 (a.u.)<sup>3</sup>
number of atoms/cell
number of atomic types
                                   18.00
number of electrons
number of Kohn-Sham states=
kinetic-energy cutoff
                                 43.0000 Ry
charge density cutoff
                                300.0000 Ry
convergence threshold
                                 1.0E-08
mixing beta
                                  0.4000
```

0 - Run your first SCF calculation for GaAs

- 0 Run your first SCF calculation for GaAs
- 1 Test convergence of the total energy against the k-point mesh

- 0 Run your first SCF calculation for GaAs
- 1 Test convergence of the total energy against the k-point mesh
- 2 Compute the equation of state (energy vs. volume):

- 0 Run your first SCF calculation for GaAs
- 1 Test convergence of the total energy against the k-point mesh
- 2 Compute the equation of state (energy vs. volume):
 - Run a sequence of SCF calculations with different lattice parameter

- 0 Run your first SCF calculation for GaAs
- 1 Test convergence of the total energy against the k-point mesh
- 2 Compute the equation of state (energy vs. volume):
 - Run a sequence of SCF calculations with different lattice parameter
 - Fit to the Murnaghan equation of state (code: ev.x) and compare to experiments

- 0 Run your first SCF calculation for GaAs
- 1 Test convergence of the total energy against the k-point mesh
- 2 Compute the equation of state (energy vs. volume):
 - Run a sequence of SCF calculations with different lattice parameter
 - Fit to the Murnaghan equation of state (code: ev.x) and compare to experiments
- 3 Compute the bands and the electronic density of states.

- 0 Run your first SCF calculation for GaAs
- 1 Test convergence of the total energy against the k-point mesh
- 2 Compute the equation of state (energy vs. volume):
 - Run a sequence of SCF calculations with different lattice parameter
 - Fit to the Murnaghan equation of state (code: ev.x) and compare to experiments
- 3 Compute the bands and the electronic density of states.
 - How large is the band gap? Does it compare well to experiments?

- 0 Run your first SCF calculation for GaAs
- 1 Test convergence of the total energy against the k-point mesh
- 2 Compute the equation of state (energy vs. volume):
 - Run a sequence of SCF calculations with different lattice parameter
 - Fit to the Murnaghan equation of state (code: ev.x) and compare to experiments
- 3 Compute the bands and the electronic density of states.
 - How large is the band gap? Does it compare well to experiments?
- 4 Repeat the band gap calculation using a hybrid functional (HSE)

- 1 Modify the GaAs input file to run SCF for Aluminum: does it work?
- 2 Use smearing to achieve SCF and test k-points convergence
- 3 Compute the equation of state (energ
 - Test the plane waves cutoff
 - Test the effect of the smearing
- 4 Compute the bands and the electron,

```
ibrav = 0,
nat = 16,
ntyp = 1,
ecutwfc = 20.d0,
ecutrho =160.0,

occupations='smearing',
degauss=0.02,
smearing='mp',
```

- 1 Modify the GaAs input file to run SCF for Aluminum: does it work?
- 2 Use smearing to achieve SCF and test k-points convergence
- 3 Compute the equation of state (energy vs. volume) + Murnhagan Fit
 - Test the plane waves cutoff
 - Test the effect of the smearing
- 4 Compute the bands and the electronic density of states

Exercise 3: Geometry of Al surface

- 1. Optimize the geometry of an Al (111) slab (4 layers thick)
- 2. Compute the band diagram and the density of states of the slab

```
&control
    calculation = 'relax'
    restart_mode='from_scratch',
    prefix='Alsurf',
    tstress = .true.
    tprnfor = .true.
    pseudo_dir = '../../Pseudo/',
    outdir='./tmp'
    forc_conv_thr = 1.d-3
/
```

```
&ions
   ion_dynamics = 'bfgs',
   ! pot_extrapolation = 'second_order',
   wfc_extrapolation = 'second_order',
/
```

Challenge:

Compute the adsorption energy of H₂ on Al(111)

Hints:

Adsorption energy:

$$E_{ads} = E(Al(111) + H_2) - E(Al(111)) - E(H_2)$$

Put H2 in atop position (see example NOT CONVERGED!!!)

Compute $E(H_2)$ in the same cell as Al(111)

Check convergence with PW cutoff and k-points

