



# Anharmonic free energy of lattice vibrations in fcc crystals from a mean-field bond lattice (using *ab-initio* potentials)

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# Motivation: Anharmonic bond distribution in bulk FCC crystals



- **Free energy** -- key quantity to determine finite temperature material properties

most popular  
approximation methods

- **Harmonic (HA)**
- **Quasi-harmonic (QHA)**
- **Thermodynamic integration (TI)**

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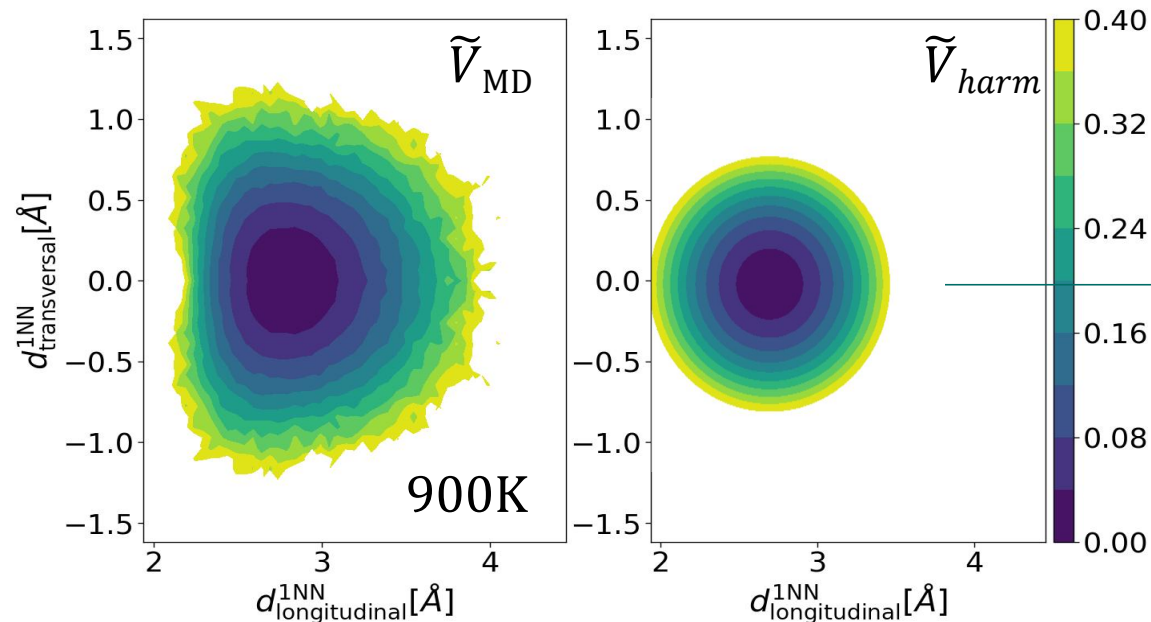


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- *Glensk et al.\** --> Finite temperature bond density -- anharmonic along the longitudinal axis



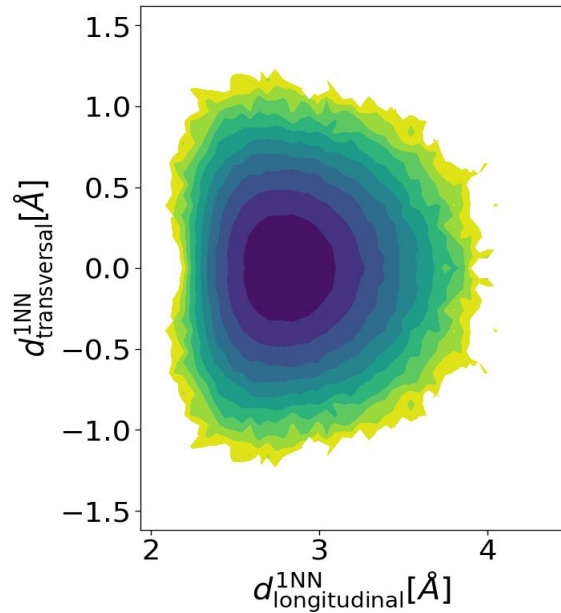
bond length,  $d^{1NN} = |\mathbf{R}_l - \mathbf{R}_i|$   $l \in [1, m]$   
effective potential,  $\tilde{V} = -k_B T \ln \rho(d^{1NN})$

HA & QHA cannot reproduce this  
anharmonic bond density

# Anharmonic bond model: Local approximation (LA)



- *Glensk et al.* --> simple anharmonic nearest-neighbor bond model --> captures bond density anharmonicity



➤ perform a few  $T = 0\text{K}$  calculations along  $\mathbf{e}_{long}$ ,  $\mathbf{e}_{t1}$  and  $\mathbf{e}_{t2}$

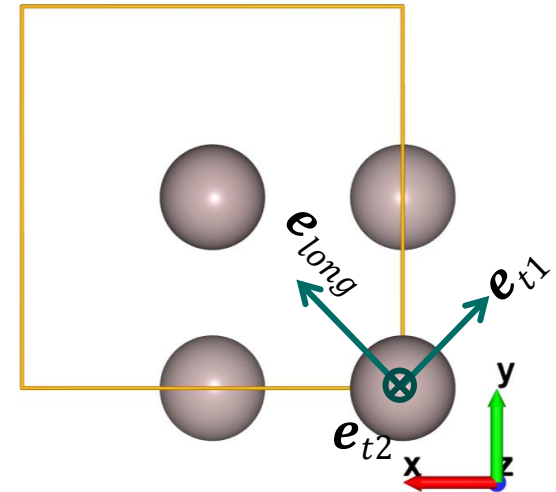
$$\text{➤ } F_a(u) = \mathbf{F}_j^{0K}(u\mathbf{e}) \cdot \mathbf{e}, \quad V_a(u) = \int_0^u F_a(u') du'$$

$$\text{➤ } E_{LA} = \frac{1}{2} \sum_j^N \sum_l^m [V_{long}(u_{lj}) + V_{t1}(u_{t1}) + V_{t2}(u_{t2})]$$

✓ meV/atom free energy accuracy to DFT!

✓ better reference for TI than QHA

✗ still requires MD trajectories



# Objectives



Build an analytical surrogate model to estimate anharmonic free energy that is

- self-consistent and robust
- requires no MD trajectories
- with a computational cost comparable to the harmonic models
- gives meV/atom free energies compared to DFT TI

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Mean-field bond lattice model\*

\*Swinburne et al., [Physical Review B, 102, 100101\(R\) \(2020\)](#)

# Bond lattice thermodynamics



- Consider a crystal with  $N$  atoms with coordinates  $\{\mathbf{R}_i\}_{i=1}^{i=N}$
- Each atom has  $m$  nearest neighbour bond vectors,

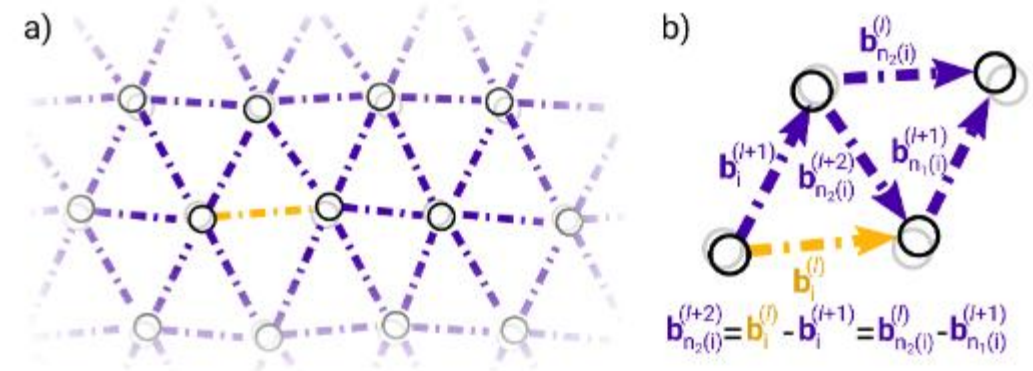
$$\mathbf{b}_i^{(l)} \equiv \mathbf{R}_{l(i)} - \mathbf{R}_i \text{ where } l \in [1, m]$$

- If we impose the following constraint:

$$\mathbf{b}_i^{(l)} - \mathbf{b}_{n_k(i)}^{(l)} = \mathbf{b}_i^{(k)} - \mathbf{b}_{n_l(i)}^{(k)}$$

which enforces zero net displacement, we get a **bond lattice**

- Total energy  $U(\{\mathbf{b}\}) = \frac{1}{2} \sum_{i=1}^N \sum_{l=1}^m V_l(\mathbf{b}_i^{(l)})$  where  $V_l(\mathbf{b}_i^{(l)})$  is a **bonding potential** (Morse, EAM etc.)





# Bond density for an FCC crystal

- At temperature  $T$ , and strain  $\epsilon$ , the total energy per atom is

$$U(\epsilon, T) = \frac{1}{2} \sum_l \int V_l(\mathbf{b}^{(l)}) \rho_{1,l}(\mathbf{b}^{(l)}, \epsilon, T) d\mathbf{b}$$

where  $\rho_{1,l}(\mathbf{b}^{(l)}, \epsilon, T)$  is the bond density and  $\tilde{V} = -k_B T \ln \rho_{1,l}(\mathbf{b}^{(l)}, \epsilon, T)$  the effective potential

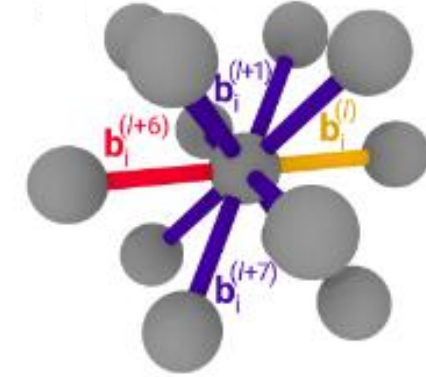




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- For an FCC crystal,  $m=12$ , all bonds are equivalent  $\Rightarrow \mathbf{a}_l^\epsilon = \mathbf{G}_{l,k} \mathbf{a}_k^\epsilon$

$\mathbf{a}_l$  = primitive lattice vector  
 $\mathbf{G}_{l,k}$  = symmetry operation

$$\triangleright V_l(\mathbf{b}^{(l)}) = V_1(\mathbf{G}_{1,l} \mathbf{b}^{(l)})$$

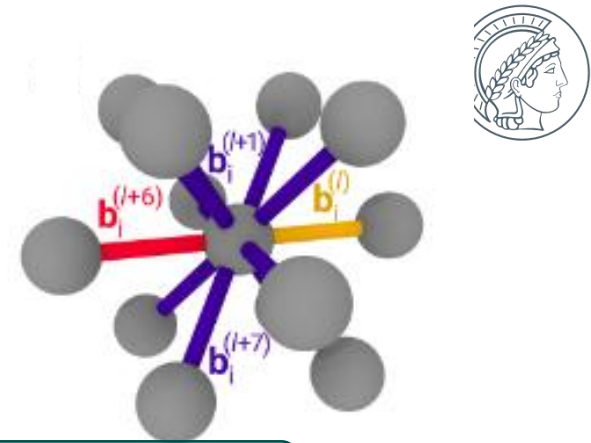
$$\triangleright \rho_{1,l}(\mathbf{b}^{(l)}, \epsilon, T) = \rho_1(\mathbf{G}_{1,l} \mathbf{b}^{(l)}, \epsilon, T)$$

and so,  $U_{fcc}^{ah}(\epsilon, T) = 6 \int V_1(\mathbf{b}) \rho_1(\mathbf{b}, \epsilon, T) d\mathbf{b} - \frac{3}{2} k_B T$  and  $F_{fcc}^{ah}(\epsilon, T) = T \int_0^T U^{ah}(\epsilon, T') / (T')^2 dT'$

# Mean-field of a bond lattice

- For bonding potential  $V_l(\mathbf{b}^{(l)}) = D(1 - \exp(-\alpha(\mathbf{b}^{(l)} - \mathbf{a}_l))) \rightarrow$  Morse potential

apply this mean-field approximation:  $\mathbf{b}_i^{(l)} - \mathbf{b}_i^{(k)} = \langle \mathbf{b}^{(l)} \rangle - \langle \mathbf{b}^{(k)} \rangle = \mathbf{a}_l^\epsilon - \mathbf{a}_k^\epsilon$



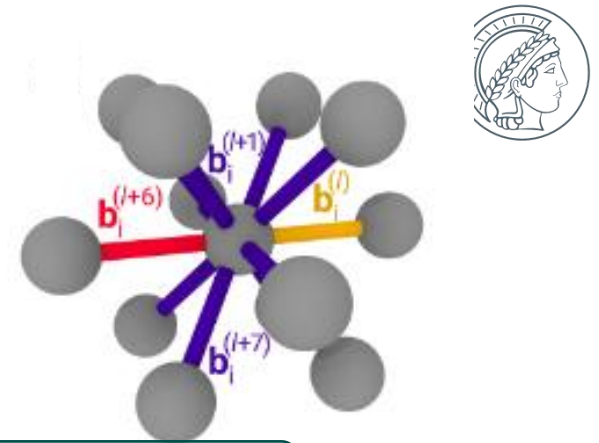
Mean-field effective potential

$$\tilde{V}_{mf}(\mathbf{b}, \epsilon) = \frac{1}{2} \sum_l^m V_l(\mathbf{b} + \mathbf{a}_l^\epsilon - \mathbf{a}_1^\epsilon) = \frac{1}{2} \sum_{l=1}^{12} V_l(\mathbf{G}_{1,l}(\mathbf{b} + \mathbf{a}_1^\epsilon) - \mathbf{a}_1^\epsilon)$$

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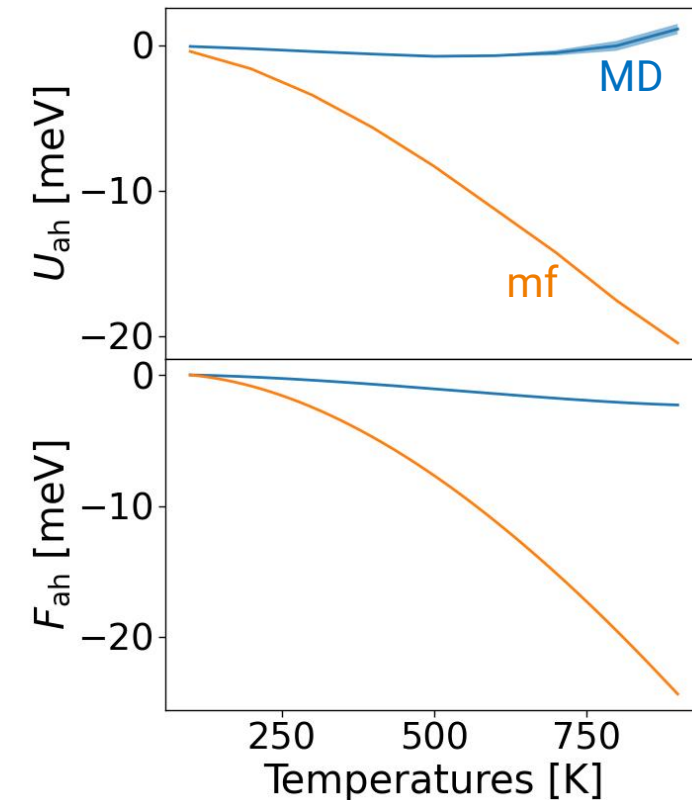
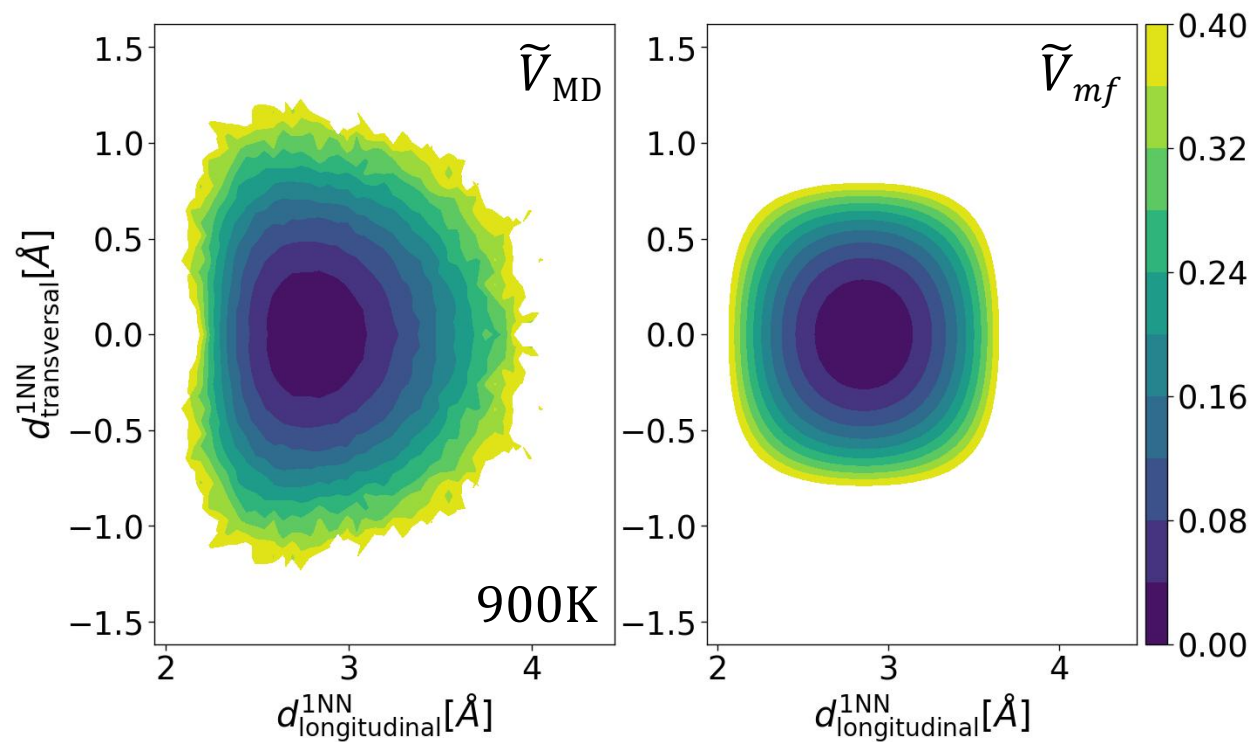
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## Mean-field effective potential

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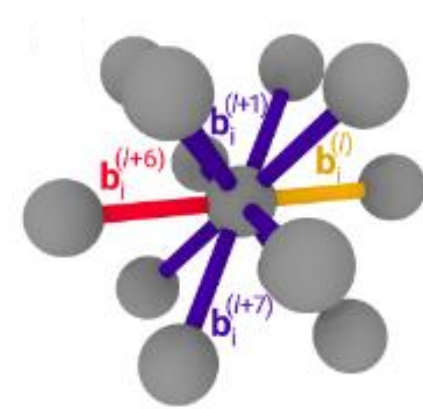
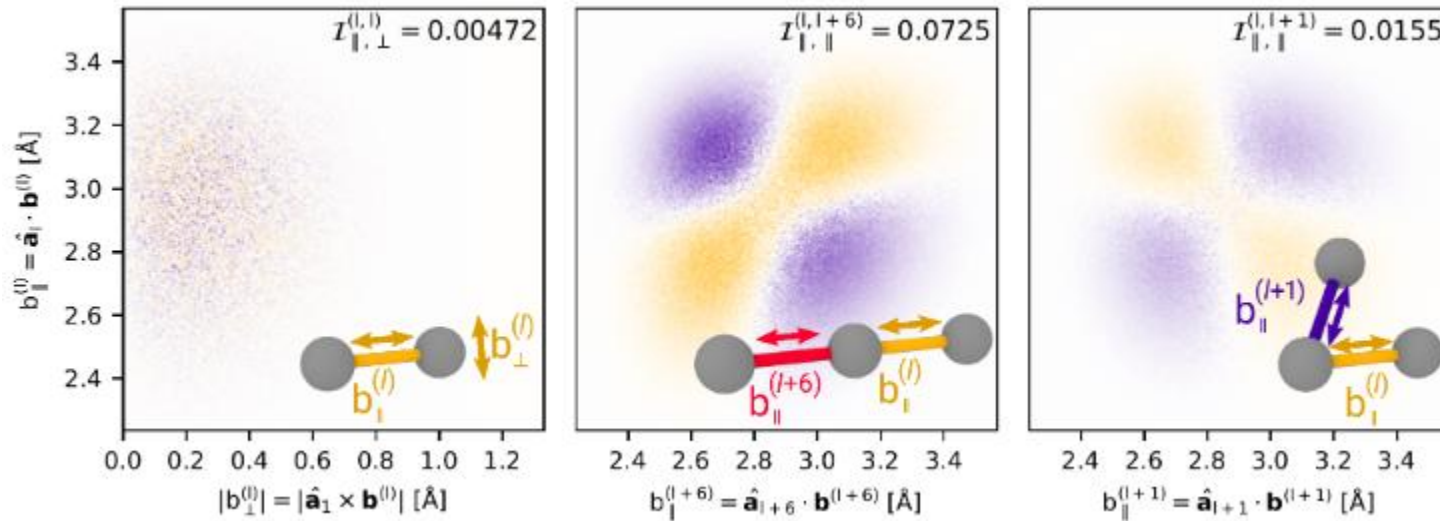




# ‘Correlated’ mean-field of a bond lattice

Bond pair correlation analysis from the MD run reveals:

- $\mathbf{b}_{\parallel}^{(l)}$ ,  $\mathbf{b}_{\perp}^{(l)}$ , are essentially uncorrelated
- $\mathbf{b}_{\parallel}^{(l)}$ ,  $\mathbf{b}_{\parallel}^{(l+6)}$ , anti-parallel bond pairs show significantly larger correlation

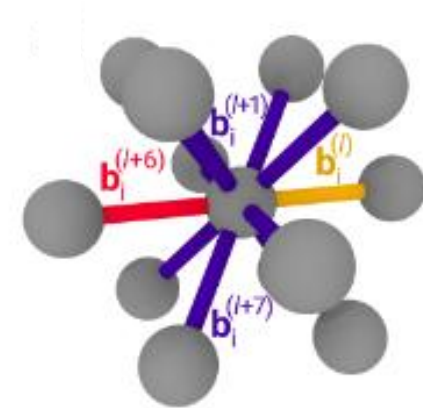
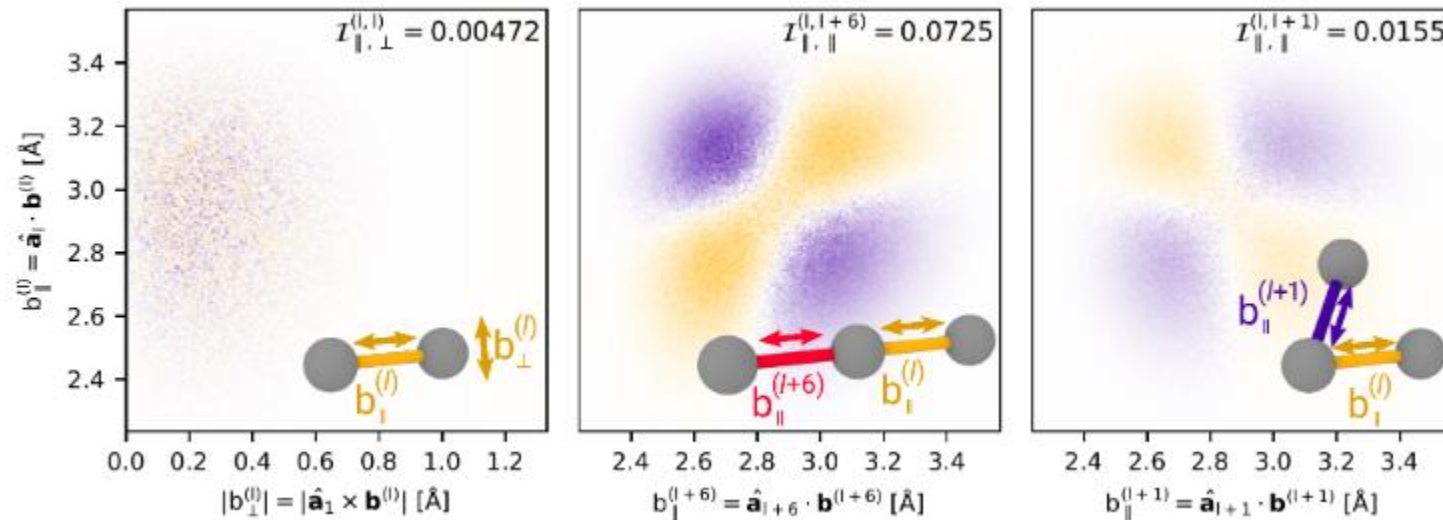




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Correlated mean-field approximation:  $\mathbf{b}_i^{(l)} - \mathbf{b}_i^{(k)} = \mathbf{a}_l^{\epsilon} - \mathbf{a}_k^{\epsilon}, \quad k \neq l$

# ‘Correlated’ mean-field model



$$\mathbf{b}_i^{(l)} - \mathbf{b}_i^{(k)} = \mathbf{a}_l^\epsilon - \mathbf{a}_k^\epsilon, \quad k \neq l$$

constraint that ensures  
 $\langle \mathbf{b} \rangle = \mathbf{a}_1^\epsilon$  at a temperature  $T$

Correlated MF effective potential

$$\tilde{V}_{mfc}(\mathbf{b}, \epsilon, T) = \tilde{V}_{mf}(\mathbf{b}, \epsilon) + \frac{1}{2} [V_1(\mathbf{b}) - V_7(\mathbf{b} - 2\mathbf{a}_1^\epsilon)] + \lambda(\epsilon, T) \hat{\mathbf{a}}_1 \cdot \mathbf{b}$$

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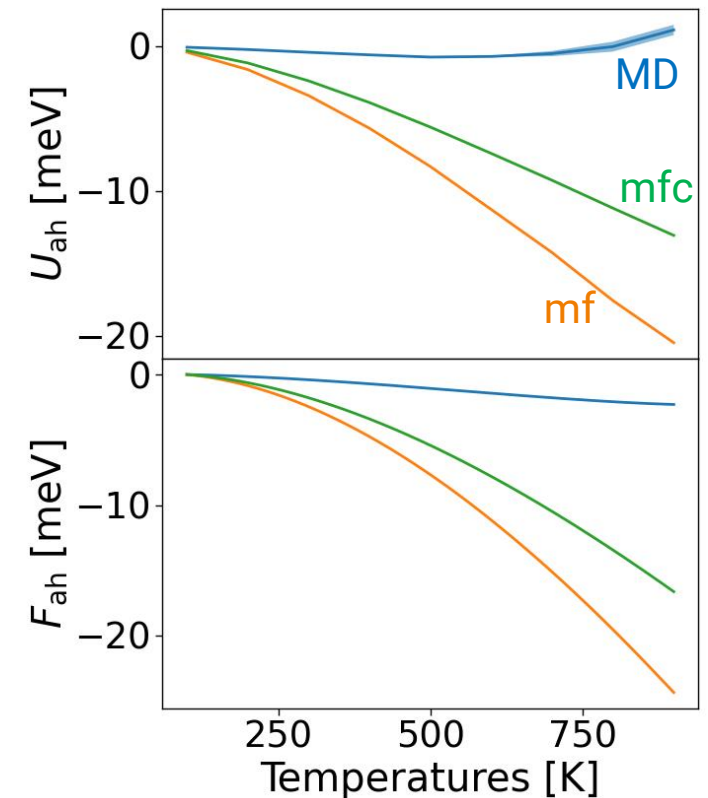
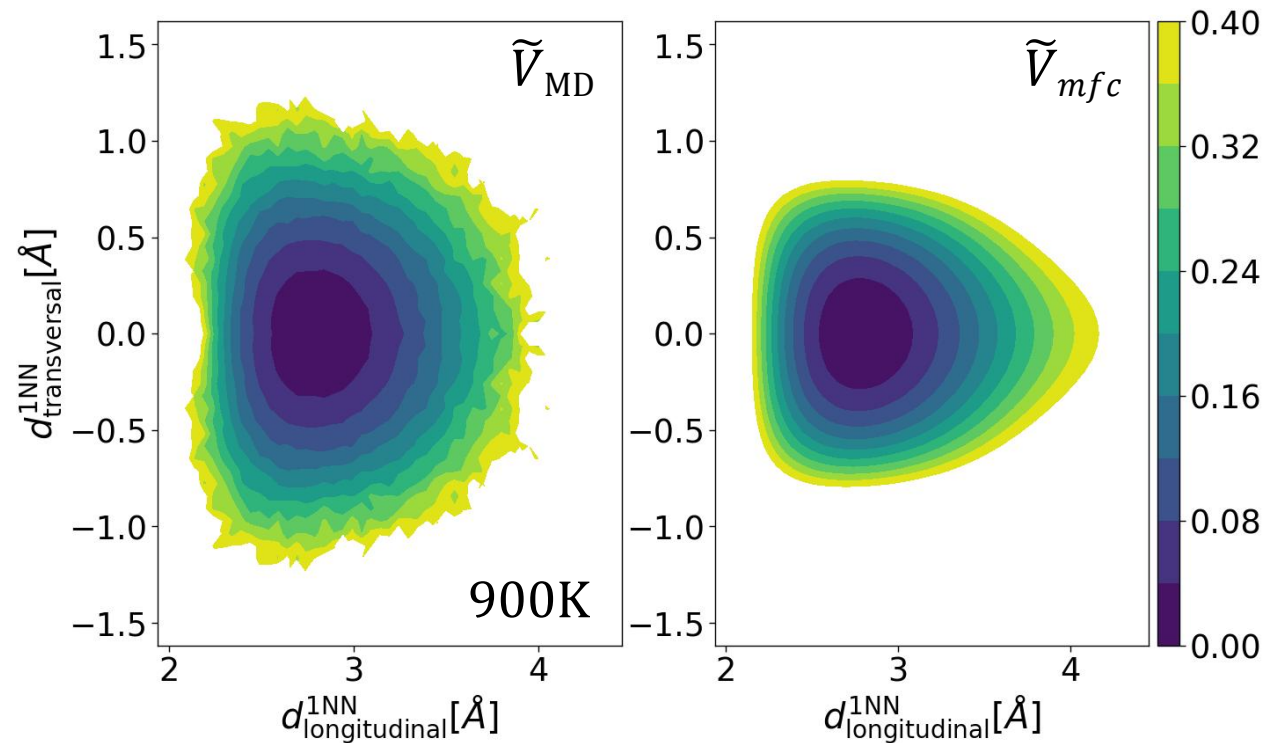


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# Virial temperature correction

Virial temperature  $T_{vir}^{mfc}(T) \equiv \frac{2}{k_B} \int (\mathbf{b} - \mathbf{a}_1^\epsilon) \cdot \nabla V_1(\mathbf{b}) \rho_1^{mfc}(\mathbf{b}, \epsilon, T) \cdot d\mathbf{b} \longrightarrow T_{vir}^{mfc}(T) \text{ may } \neq T$

- optimize an effective temperature  $\tilde{T}$ :  $T_{vir}^{mfc}(\tilde{T}) = T$
- virial temperature corrected MFC effective potential  $\tilde{V}_{mfcv}(\mathbf{b}, \epsilon, T) = \tilde{V}_{mfc}(\mathbf{b}, \epsilon, \tilde{T})$



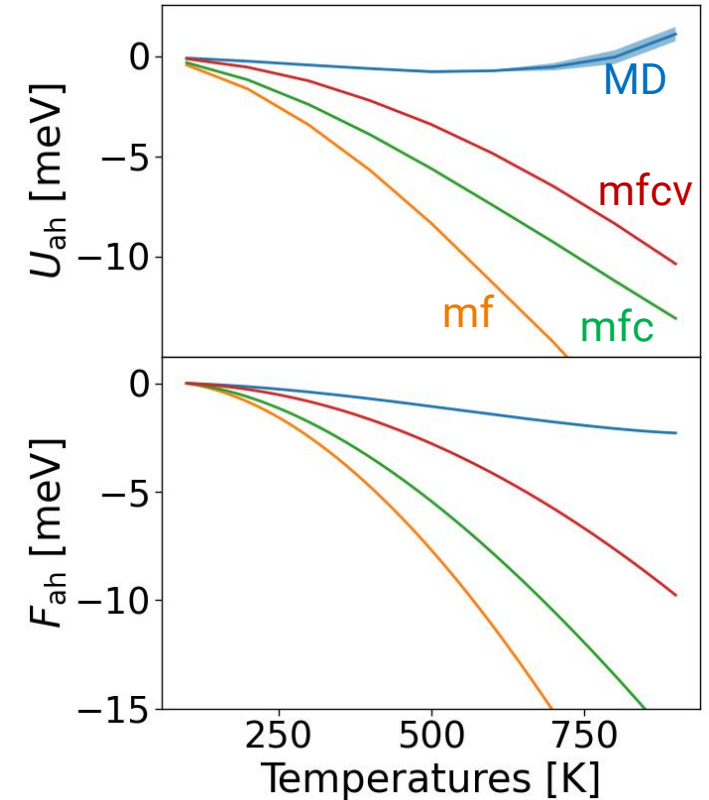
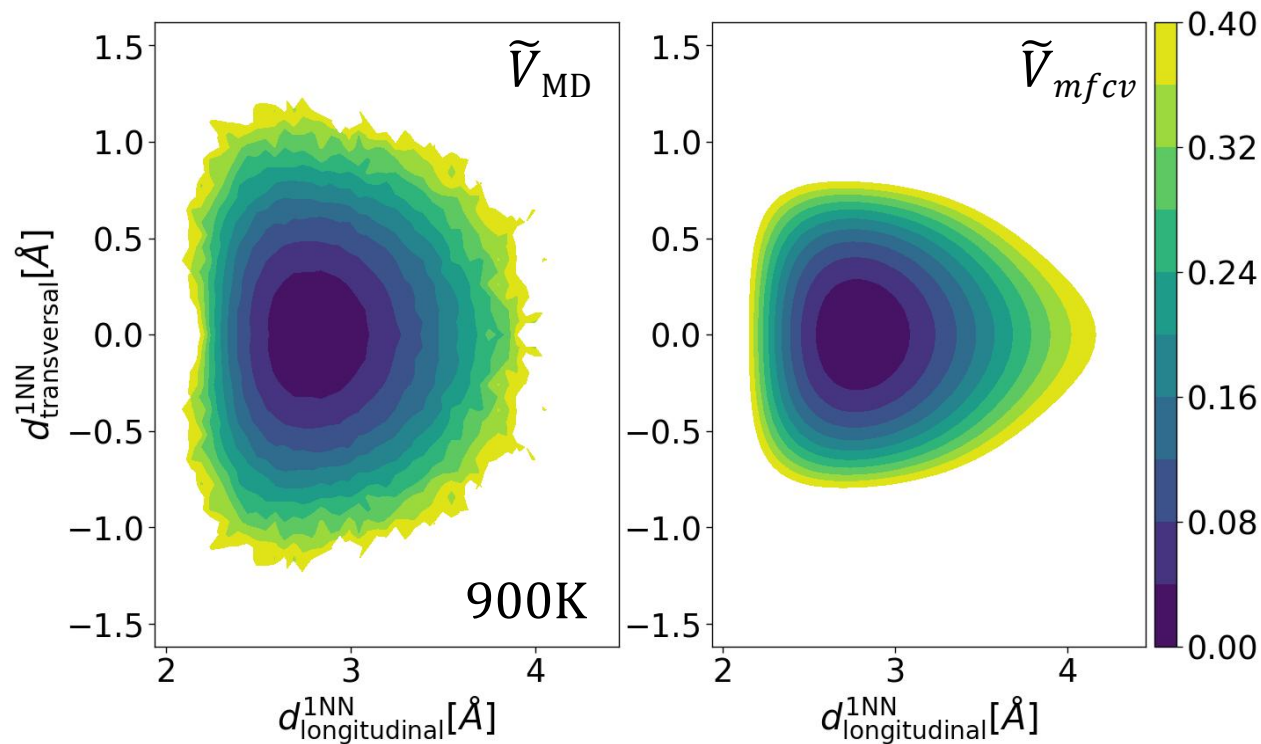
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# Zero virial pressure -- finite strain

Virial pressure  $P_{vir}^{mfc}(T, \epsilon) \equiv \frac{\rho_0}{(1+\epsilon)^d} \left( k_B T + \frac{m}{d} \langle \mathbf{a}_1^\epsilon \cdot \nabla V_1 \rangle \right)$

- Further approximate that the model is formally complete when virial pressure at  $\tilde{T}$  is 0.

- finite strain mean-field approximation  $\mathbf{b}_i^{(l)} - \mathbf{b}_i^{(k)} = (\mathbf{a}_l^\epsilon - \mathbf{a}_k^\epsilon)(1 + \alpha), \quad k \neq l$

such that  $\alpha(\tilde{T}) = \epsilon(\tilde{T})$  when  $P_{vir}^{mfc}(\tilde{T}, \epsilon) = 0$

- Mean-field effective potential changes to  $\tilde{V}_{mf}(\mathbf{b}, \epsilon, \tilde{T}) = \frac{1}{2} \sum_l^m V_l(\mathbf{b} + (\mathbf{a}_l^\epsilon - \mathbf{a}_1)(1 + \alpha(\tilde{T})))$

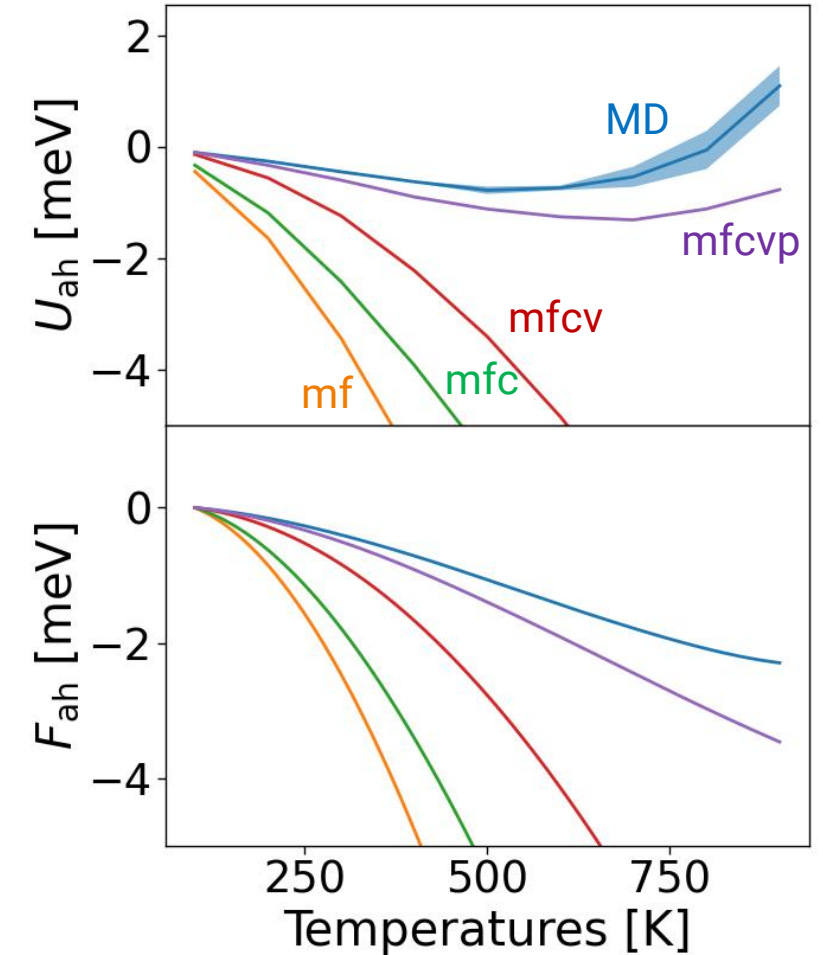
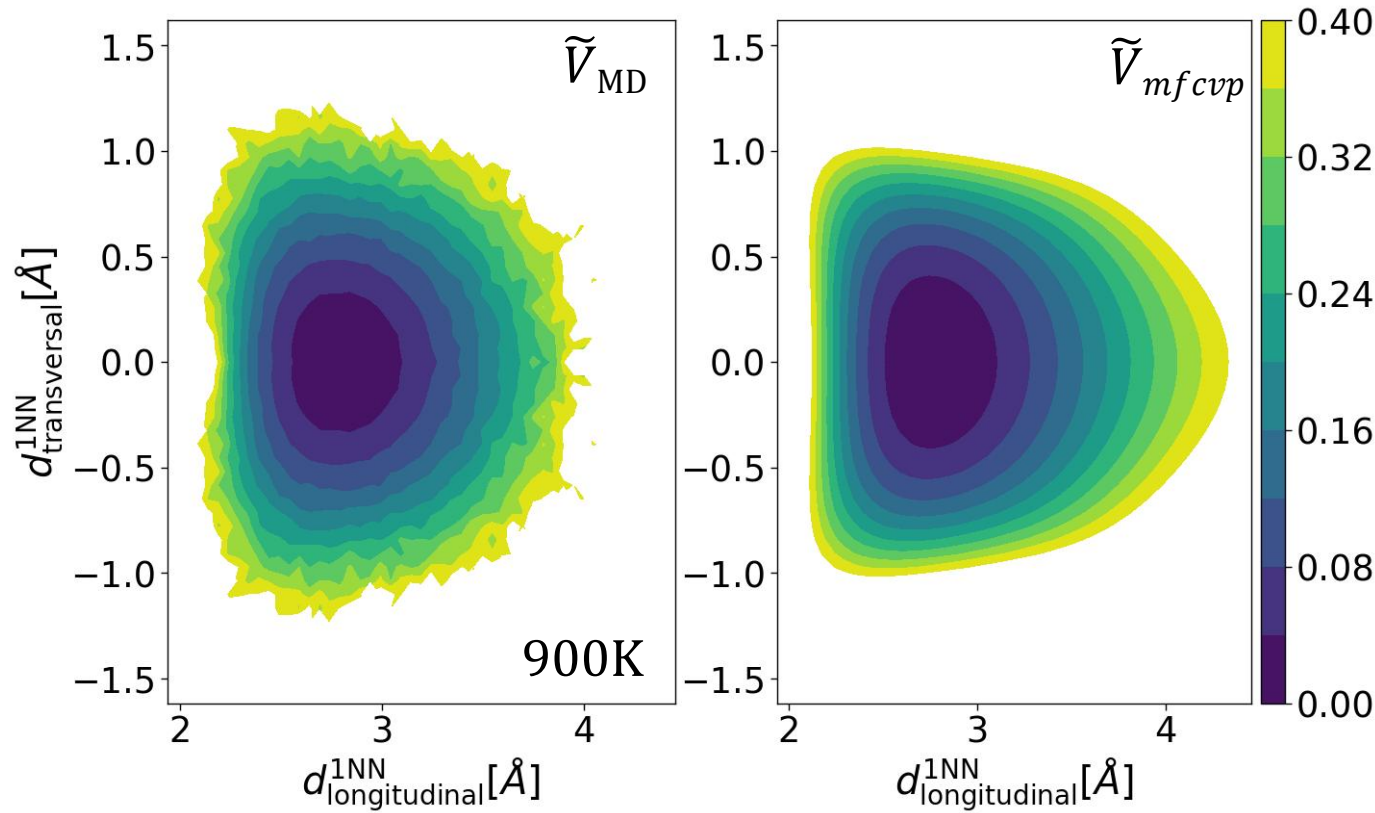
New optimization: for any  $T$ , find  $\tilde{T}$ ,  $\alpha(\tilde{T})$  while ensuring:

- $P_{vir}^{mfc}(\tilde{T}, \epsilon) = 0$
- $T_{vir}^{mfc}(\tilde{T}, \epsilon, \alpha(\tilde{T})) = T$
- $\alpha(\tilde{T}) = \epsilon(\tilde{T})$

# Zero virial pressure, temperature corrected model



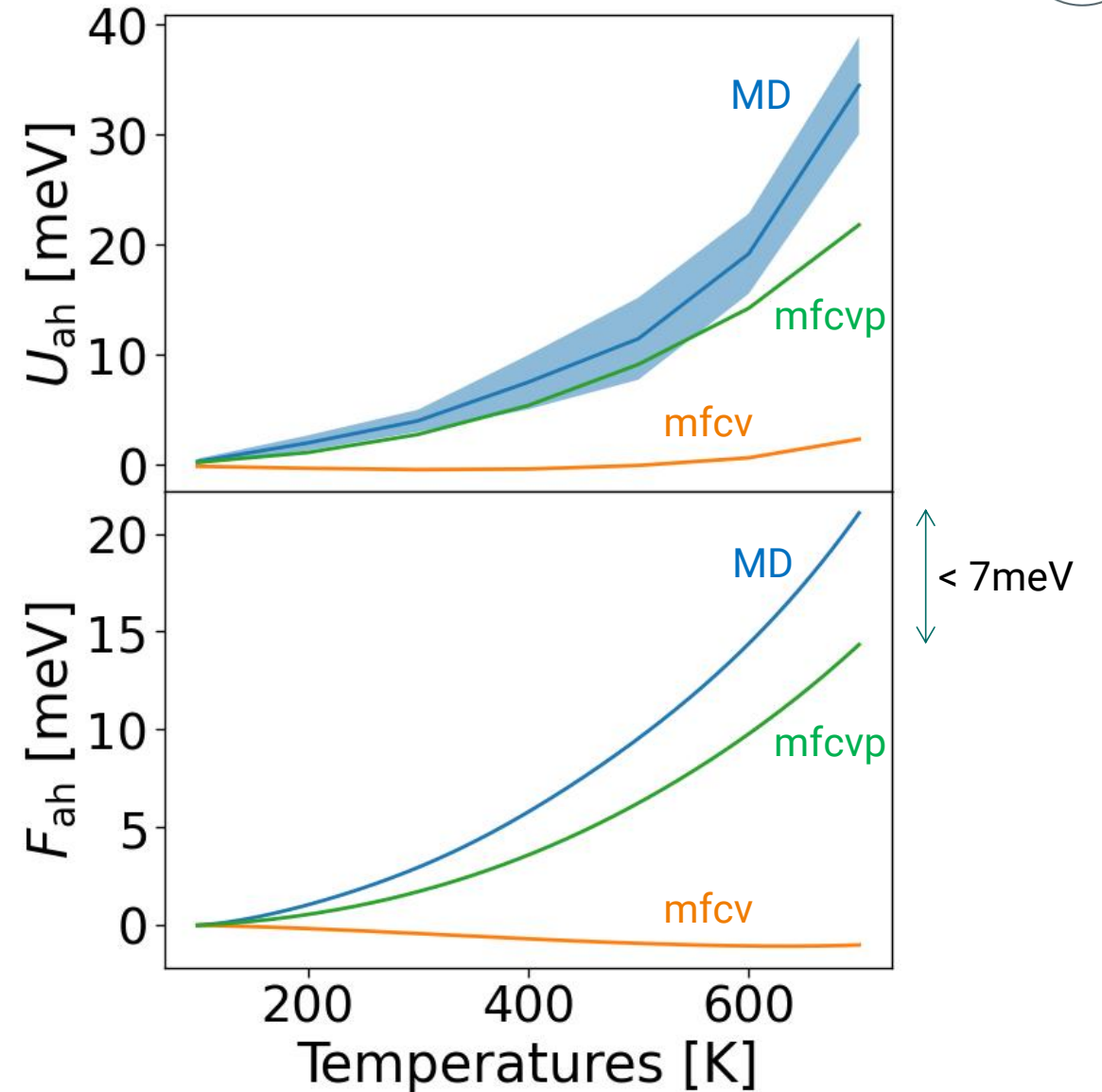
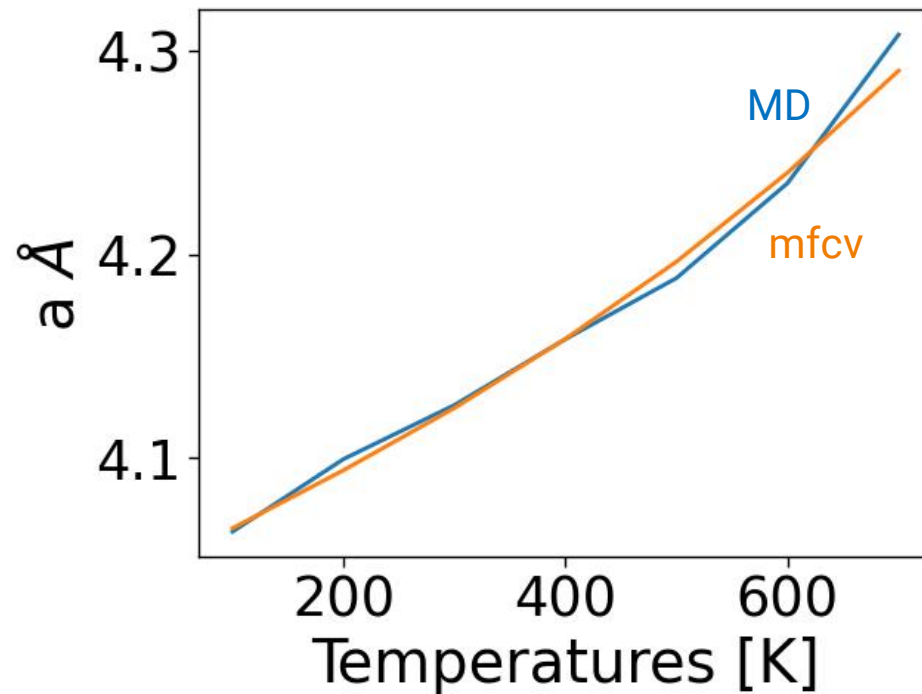
Full virial corrected MFC effective potential:  $\tilde{V}_{mfcvp}(\mathbf{b}, \epsilon, T) = \tilde{V}_{mfc}(\mathbf{b}, \epsilon, \tilde{T}, \alpha(\tilde{T}))$



# Mean-field bond lattice model vs. zero pressure MD



For  $P_{vir}^{mfc}(\tilde{T}, \epsilon) = 0$ ,



# Extension to an *ab-initio* potential

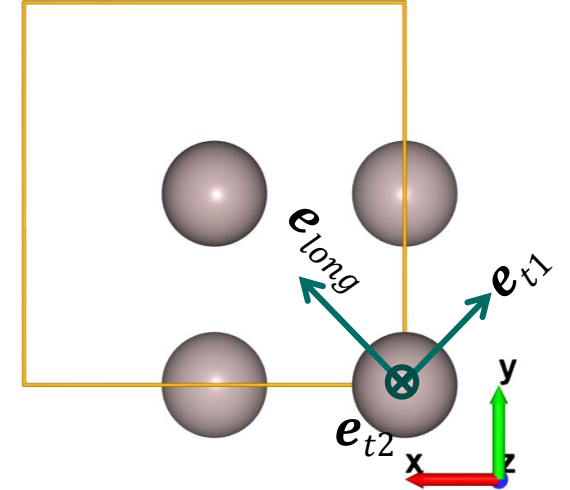
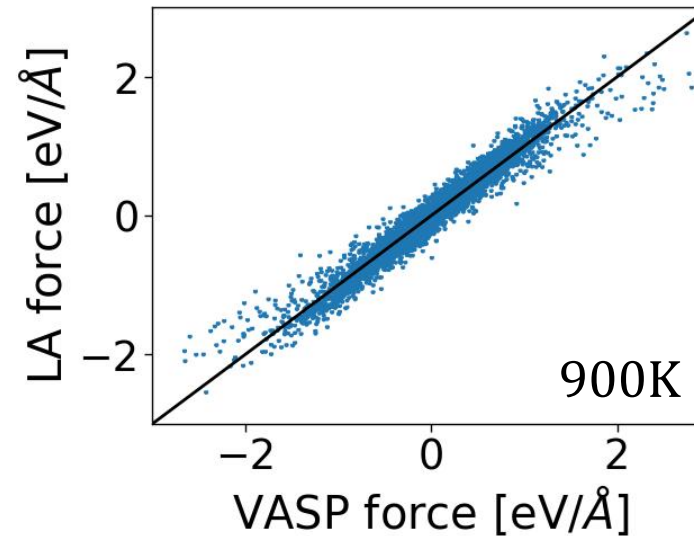


System: PBE Aluminium

Local approximation bonding potential  $V_1(\mathbf{b}) = V_{long}(u_{lj}) + V_{t1}(u_{t1}) + V_{t2}(u_{t2})$

➤ perform few  $T = 0\text{K}$  calculations along  $\mathbf{e}_{long}$ ,  $\mathbf{e}_{t1}$  and  $\mathbf{e}_{t2}$

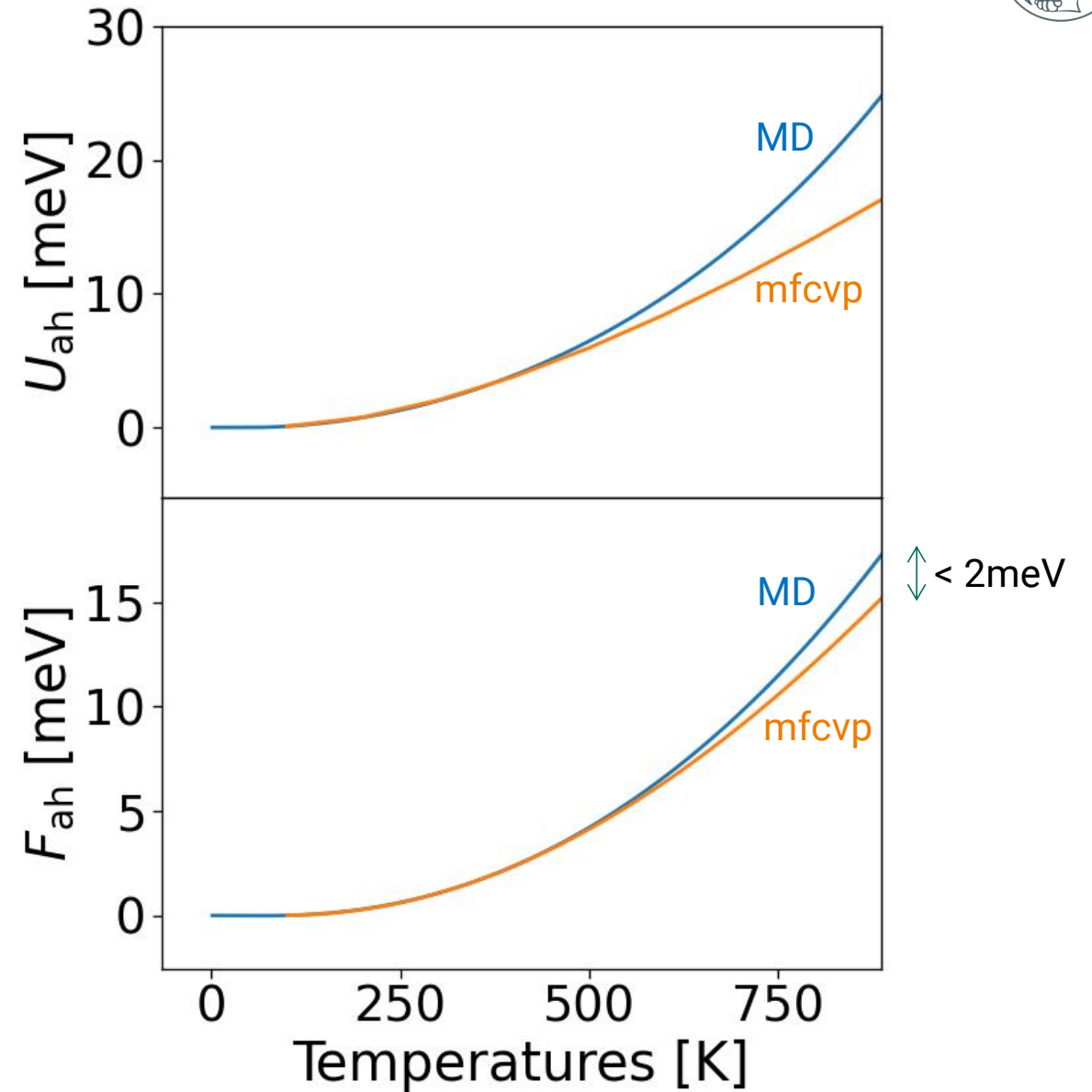
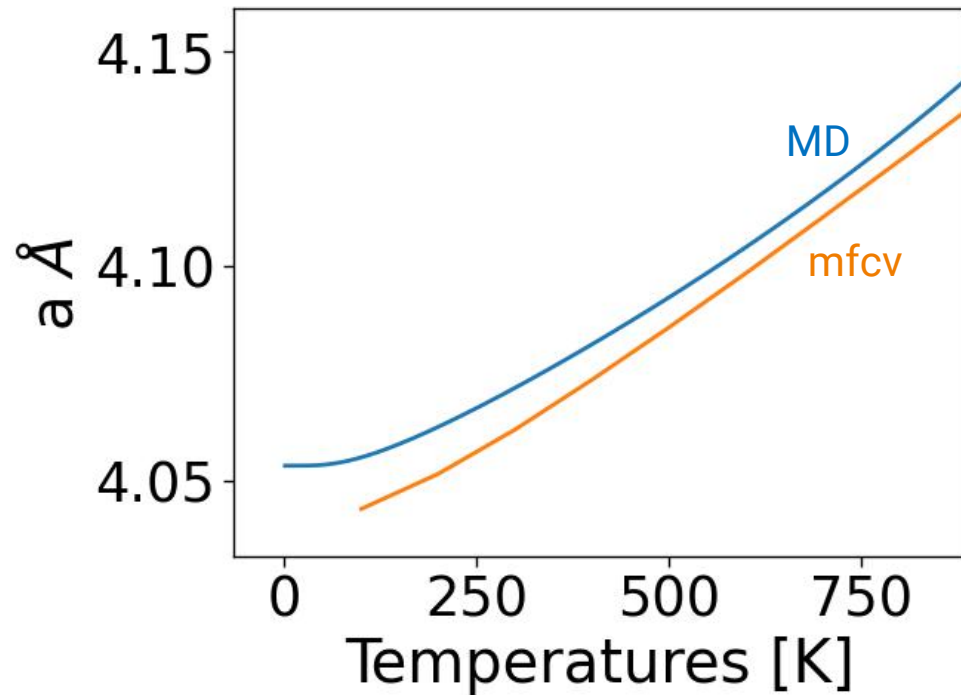
➤  $F_a(u) = \mathbf{F}_j^{0K}(u\mathbf{e}) \cdot \mathbf{e}$ ,  $V_a(u) = \int_0^u F_a(u') du'$



# Mean-field bond lattice model vs. TI

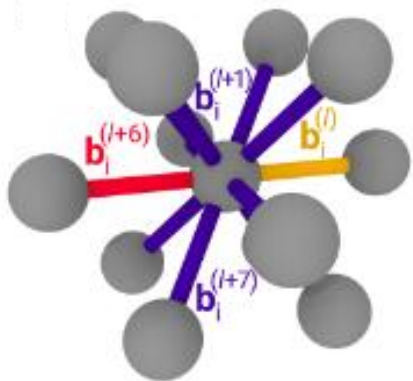


For  $P_{vir}^{mfc}(\tilde{T}, \epsilon) = 0$ ,



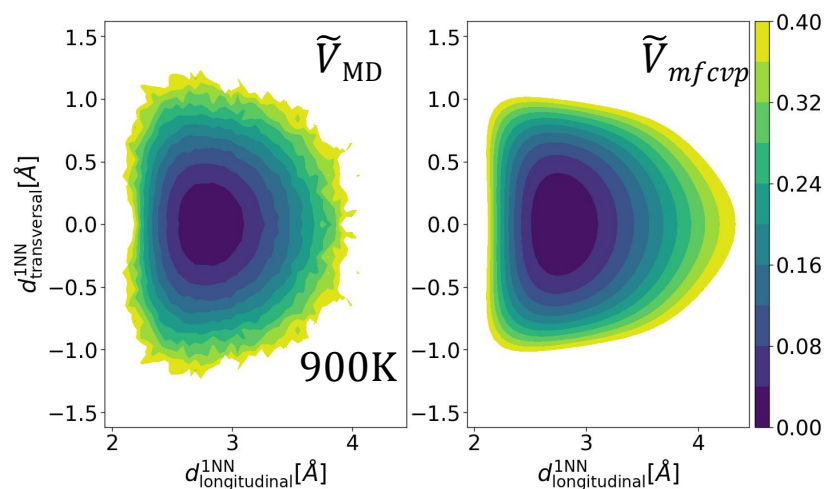
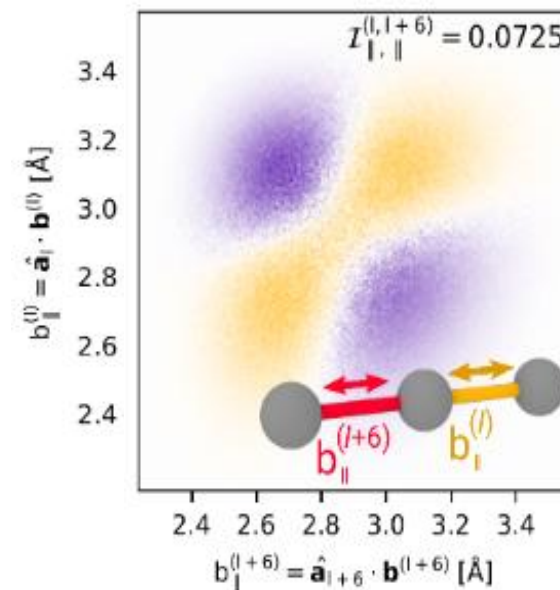


# Summary



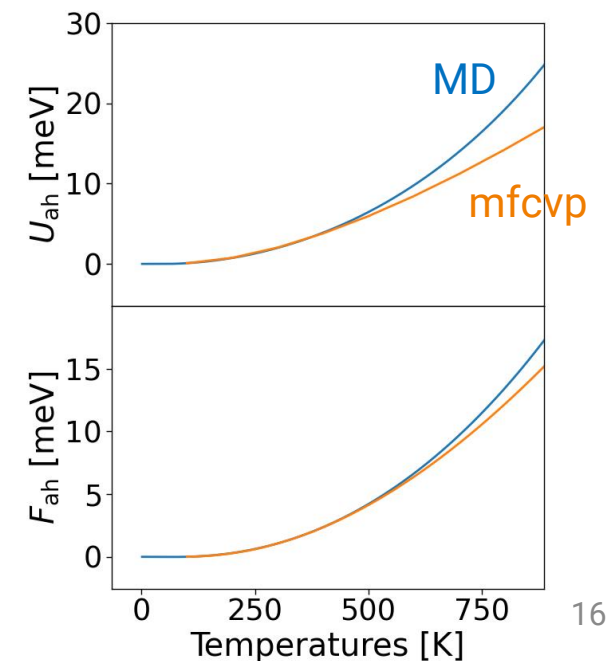
analytical self-consistent mean-field bond lattice model...

...accounts for bond pair correlations, virial corrections...



...to predict anharmonic bond distribution...

...and the anharmonic free energy



**Thank you for your  
attention!**

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