

5 External-field-driven flow

One of the simplest ways to implement time-dependent boundary conditions in SOMA is to apply a time-dependent external field that is only nonzero close to the boundaries and has the form:

$$E_i(\mathbf{r}) = f_i(\mathbf{r})\phi_i(\mathbf{r}), \quad \text{where is the stress dependence} \quad (5.1)$$

where $i = A, B$ denotes the monomer types. In this section, self-assembled lamellae of diblock-copolymers are moved at constant speed v perpendicular to their orientation using spatially periodic external fields close to the boundaries. In a system that moves along with the external field, each monomer experiences a friction force ζv in the opposite direction. This force, which is a consequence of the applied MC scheme to simulate Brownian dynamics, causes the lamellae to bend and eventually tear at high values of v .

5.1 Reference system

A system of $n = 750$ symmetric diblock-copolymers with $N_A = N_B = N/2 = 16$ and $\chi N = 20$ is used. The box dimensions are $L_x \times L_y \times L_z = 2.5 \times 2.82 \times 1 R_e^3$, which corresponds to $\sqrt{N} = 106$. The spatial discretisations are $\Delta x = 1/16 R_e$, $\Delta y = 47/800 R_e$ and $\Delta z = 1 R_e$. To generate the initial lamellar structure, external fields are applied, as shown in Figure 5.1. The interlayer spacing of $d = 1.41 R_e$ was found to be stable over the duration of the simulations, but it does not correspond to the equilibrium spacing.

Subsequently, the external fields are switched off everywhere except at a distance less than $b = 0.5 R_e$ from the boundaries in the x -direction, so the length of the part of the lamellae that is not supported by the external fields is $L = 1.5 R_e$. Every Δt MCS, the fields are moved by a distance of Δy in the y -direction, so the velocity is

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define a Péclet number $Pe = \frac{v R_e}{D}$ and discuss significance

$v = 47 R_e / (800 \Delta t)$. The external fields balance the friction forces at the boundaries and therefore act as bearings for the lamellae.

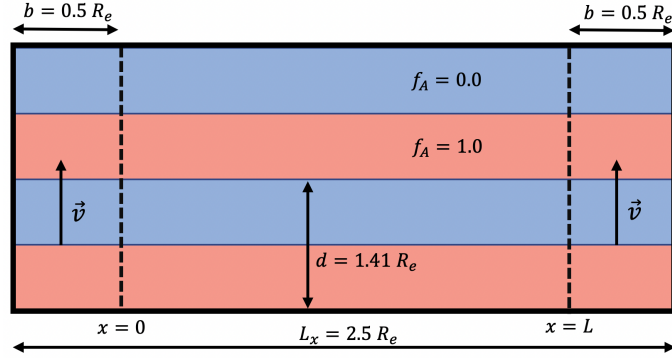


Figure 5.1: Sketch of the external field $f_A(\mathbf{r})$. Red domains correspond to $f_A = 1.0$, blue domains to $f_A = 0.0$. $f_B(\mathbf{r})$ is exactly complementary. Within the region bounded by the dotted lines, the external fields are switched off after the initial lamella structure has been generated.

5.2 Bending modulus

The free energy for a bent monolayer of diblock-copolymers is [25]:

Sollte hier nicht auch ein Integral über L_y und L_z sein – die freie Energie ist extensiv

$$F = \frac{1}{2} K \int dx \left(\frac{\partial^2 u}{\partial x^2} \right)^2 \equiv \int dx f(u''), \quad \text{with } K \sim [K] \left(\frac{R}{R_e} \right)^2 \quad (5.2)$$

wo K ist die Einheit von K

where K is the bending modulus and $u \equiv u(x)$ is the deformation profile of the lamella center of mass. The total friction force acting on the monolayer is:

$$F_{fric} = -\rho_0 d L L_z \zeta v = -\frac{nN}{L_x L_y} L d \zeta v, \quad (5.3)$$

mit anderen Symbolen verwenden, um es nicht mit F für freie Energie zu verwechseln

where $\rho_0 = nN/V$ is the mean monomer density in the system. The deformation profile can be obtained by solving the following Euler-Lagrange equation:

mit zwei Gleichungen lösen

$$\frac{d^2}{dx^2} \left(\frac{\partial f}{\partial u''} \right) = F_{fric}. \quad (5.4)$$

Defining $q \equiv -\frac{nN}{L_x L_y} d \zeta v$, this gives:

$$K \frac{\partial^4 u}{\partial x^4} = q. \quad (5.5)$$

With the boundary conditions $u(0) = u(L) = 0$ and $u''(0) = u''(L) = 0$, one obtains:

$$u(x) = \frac{qx}{24K} (L^3 - 2L^2x + x^3), \quad (5.6)$$

in complete analogy to a beam bending under a uniform load q in the Euler-Bernoulli theory. The resulting lamella profile for $v = 0.34R_e/\tau_R$ is shown in Figure 5.2. The fit is in excellent agreement with the simulation data.

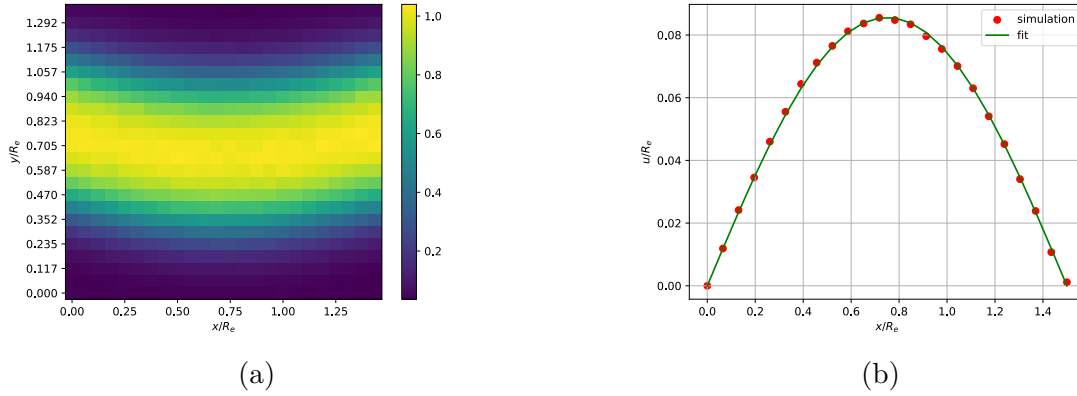


Figure 5.2: (a) Heatmap of the steady-state lamella profile in the reference frame that moves with the external field, averaged over all lamellae. (b) Lamella center of mass curve for $v = 0.34R_e/\tau_R$. The fit corresponds to (5.6).

The maximum deflection is:

$$u_{max} = u(L/2) = \frac{5qL^4}{384K}. \quad (5.7)$$

To obtain the bending modulus, u_{max} is measured for various values of q , this is shown in Figure 5.3. From (5.7), one obtains $K = 39.00 k_B T R_e$. In equilibrium, it may also be obtained directly from the simulation parameters [25]:

$$K = \frac{3}{16} \left(\frac{12}{\pi^2} \right)^{1/3} \left(\frac{\gamma}{k_B T} \right)^{4/3} \bar{N}^{-1/6} R_e^{5/3} k_B T, \quad (5.8)$$

$\left(\frac{1}{\bar{N}} \right)^{4/3} \bar{N}^{-1/6} R_e^{5/3} k_B T = \left(\frac{1}{\bar{N}} \right)^{4/3} R_e^{-1/6} \bar{N}^{-1/6} R_e^{5/3} k_B T = \left(\frac{1}{\bar{N}} \right)^{4/3} \frac{1}{\bar{N}} k_B T$
dimensionless

where γ is the interfacial tension. In the strong segregation limit, it reads [21]:

$$\frac{\gamma R_e^2}{\sqrt{N} k_B T} = \sqrt{\frac{\chi N}{6}} \left(1 - \frac{4 \ln 2}{\chi N} \right). \quad (5.9)$$

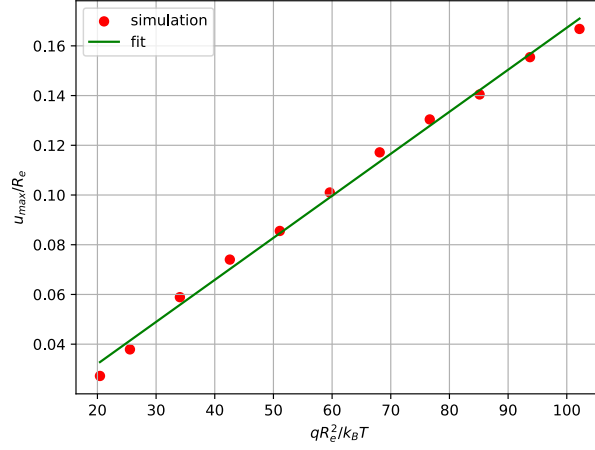


Figure 5.3: Maximum deflection u_{max} as a function of the applied load q . The fit corresponds to (5.7).

The result is $K = 38.94 k_B T R_e$, in excellent agreement with the simulation results. However, it should be noted that (5.8) is derived from the equilibrium spacing d_0 . For $\chi N = 20$, [25] predicts $d_0 = 1.24 R_e$, which is too small (**Need some reference value here to cite**). This is likely to indicate that the strong segregation is not fully reached at $\chi N = 20$. Nevertheless, the results might suggest that the bending modulus K only weakly depends on the monolayer thickness.

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