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# SNAP: Spectral neighbor analysis method for automated generation of quantum-accurate interatomic potentials for LAMMPS

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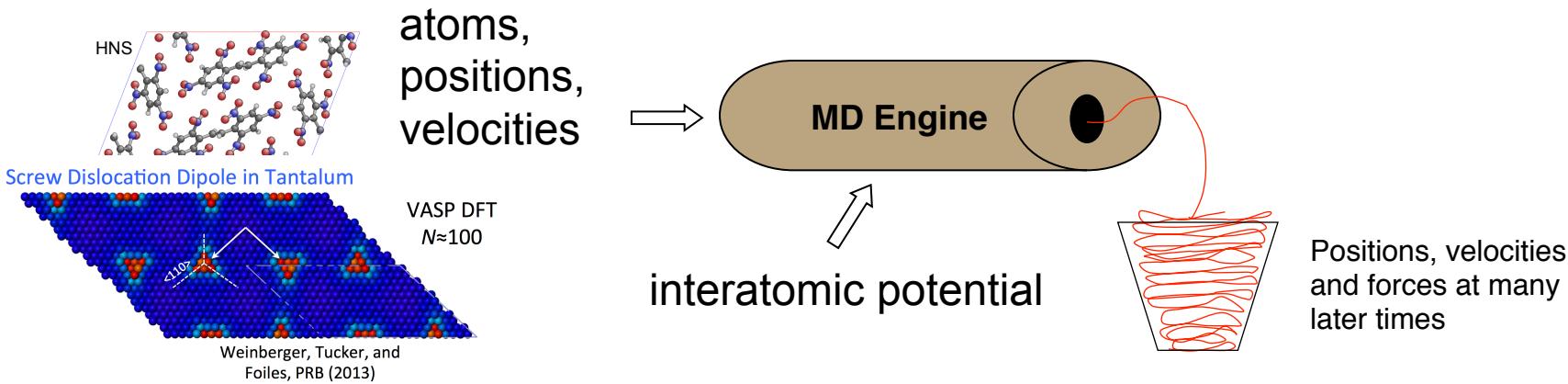
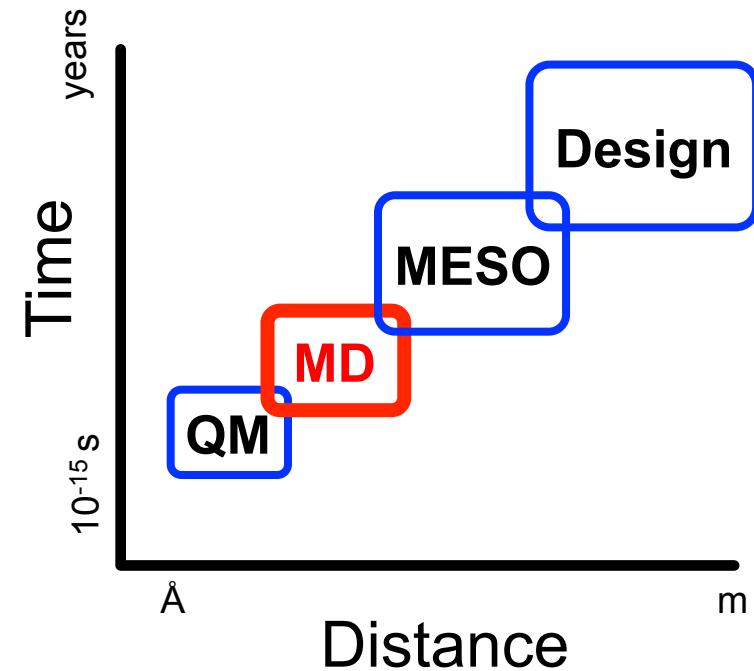
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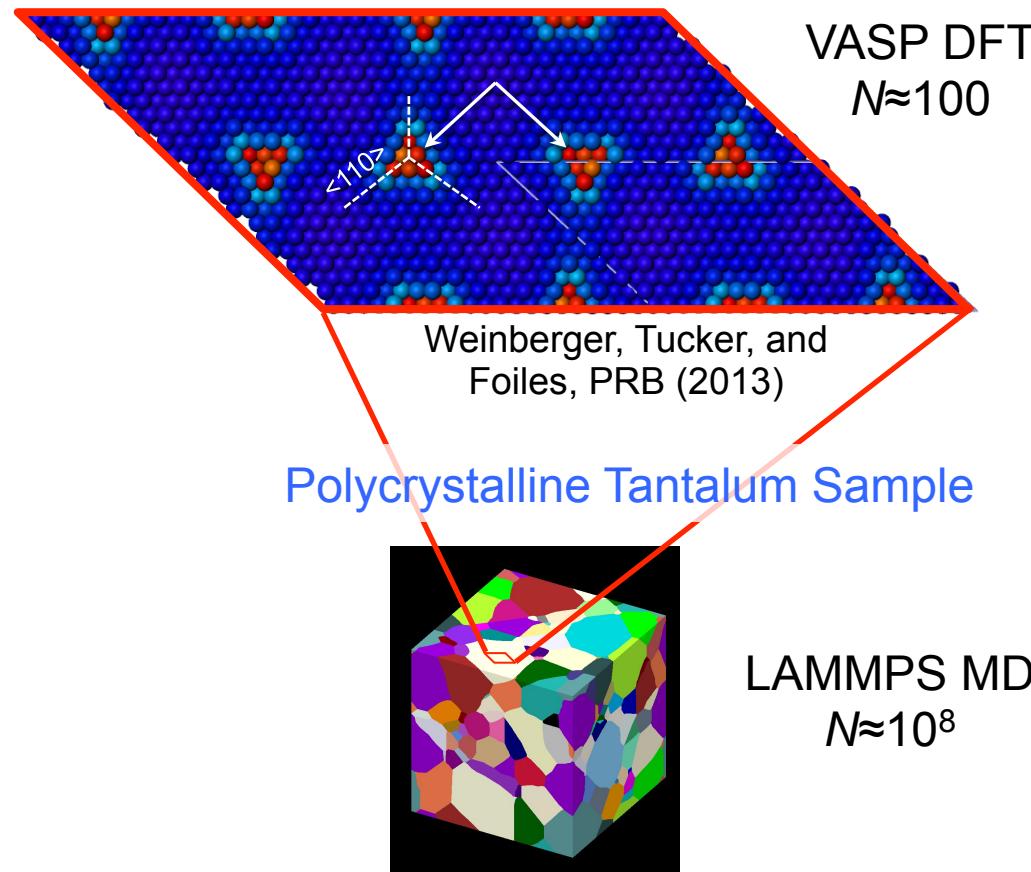
# Why Use Molecular Dynamics Simulation

- Continuum models require underlying models of the materials behavior
- Quantum methods can provide very complete description for 100s of atoms
- Molecular Dynamics acts as the “missing link”
  - Bridges between quantum and continuum models
  - Moreover, extends quantum accuracy to continuum length scales; retaining atomistic information

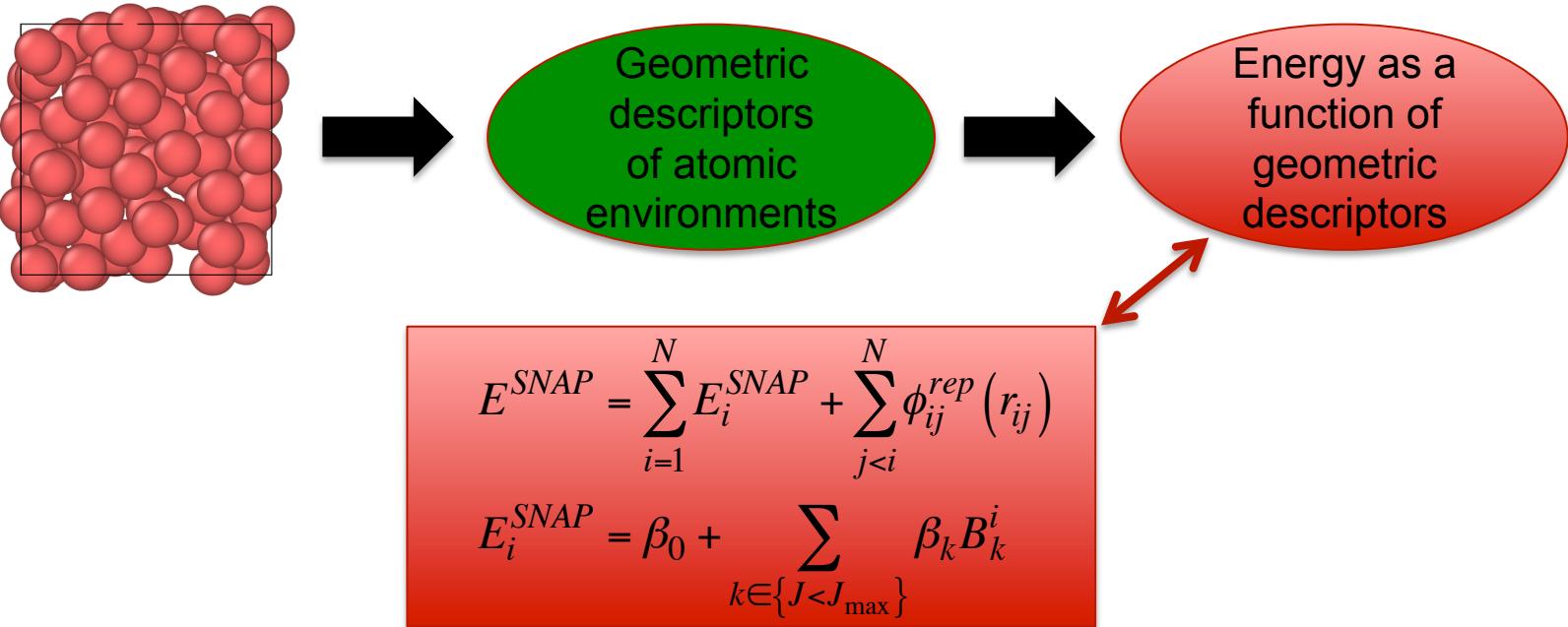


# Example: Plasticity in BCC Metals

## Screw Dislocation Motion in BCC Tantalum



# SNAP: Spectral Neighbor Analysis Potentials



- **GAP (Gaussian Approximation Potential)**: Bartok, Csanyi et al., *Phys. Rev. Lett.*, 2010. Uses 3D neighbor density bispectrum and **Gaussian process regression**.
- **SNAP (Spectral Neighbor Analysis Potential)**: Our SNAP approach uses GAP's neighbor bispectrum, but replaces Gaussian process with **linear regression**.
  - More robust
  - Lower computational cost
  - Decouples MD speed from training set size
  - Enables large training data sets, more bispectrum coefficients
  - Straightforward sensitivity analysis

# Bispectrum Components as Descriptor

- Neighbors of each atom are mapped onto unit sphere in 4D

$$(\theta_0, \theta, \phi) = \left( \theta_0^{\max} r/r_{cut}, \cos^{-1}(z/r), \tan^{-1}(y/x) \right)$$

- Expand density around each atom in a basis of ***4D hyperspherical harmonics***,
- Bispectrum components of the 4D hyperspherical harmonic expansion are used as the geometric descriptors of the local environment
  - Preserves universal physical symmetries
  - Rotation, translation, permutation
  - Size-consistent

$$u_{m,m'}^j = U_{m,m'}^j(0,0,0) + \sum_{r_{ii'} < R_{cut}} f_c(r_{ii'}) w_i U_{m,m'}^j(\theta_0, \theta, \phi)$$

$$B_{j_1, j_2, j} = \sum_{m_1, m'_1 = -j_1}^{j_1} \sum_{m_2, m'_2 = -j_2}^{j_2} \sum_{m, m' = -j}^j (u_{m,m'}^j)^* H_{j_1 m_1 m'_1 \atop j_2 m_2 m'_2}^{j m m'} u_{m_1, m'_1}^{j_1} u_{m_2, m'_2}^{j_2}$$

Symmetry relation:  $\frac{B_{j_1, j_2, j}}{2j+1} = \frac{B_{j, j_2, j_1}}{2j_1+1} = \frac{B_{j_1, j, j_2}}{2j_2+1}$

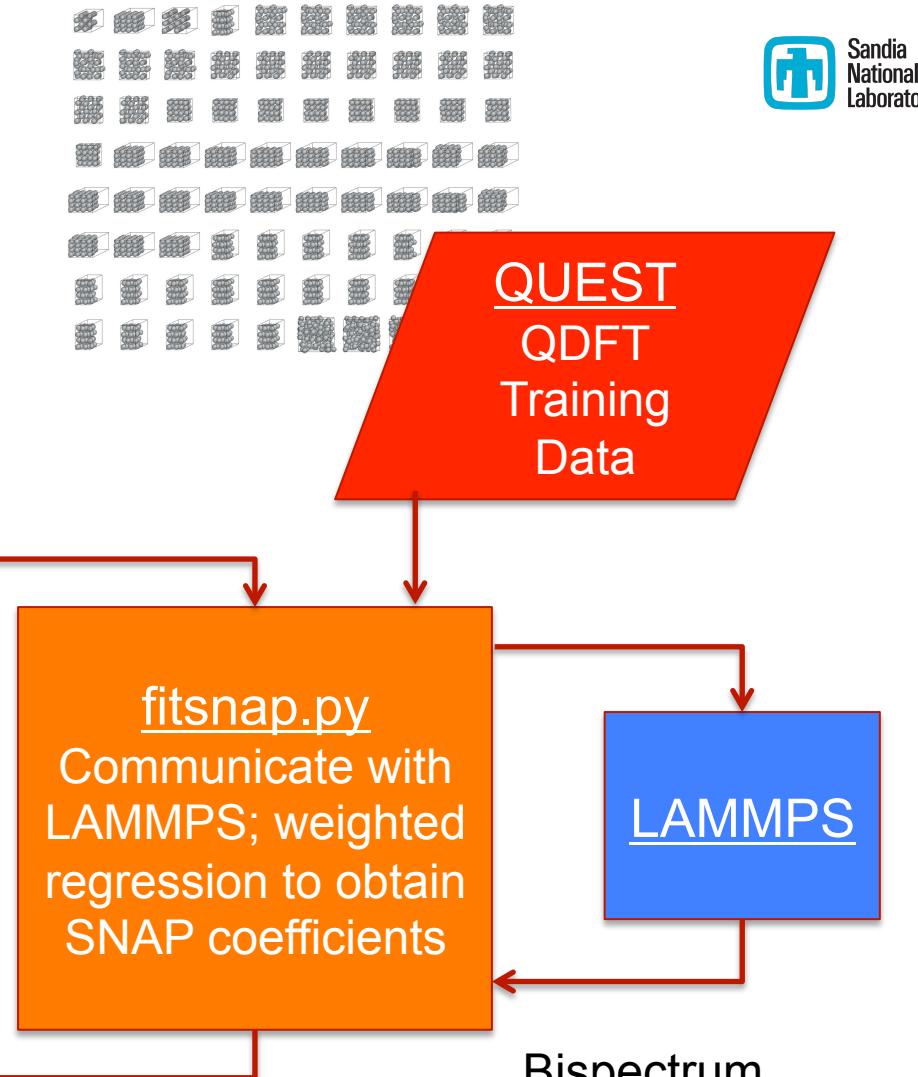
# SNAP Fitting Process

## FitSnap.py

- “Hyper-parameters”
- Cutoff distance
  - Group Weights
  - Number of Terms
  - Etc.



- Metrics
- Force residuals
  - Energy residuals
  - Elastic constants
  - Etc.

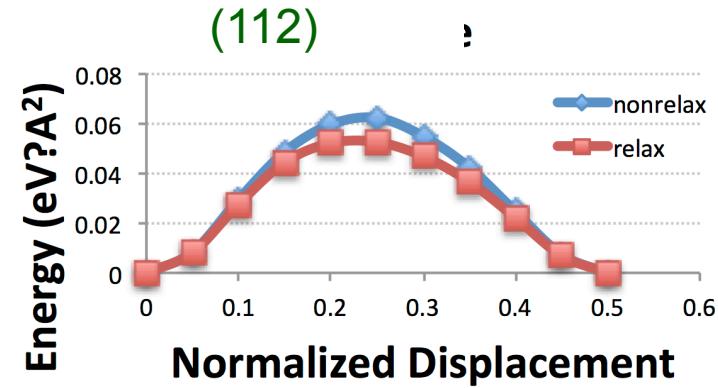
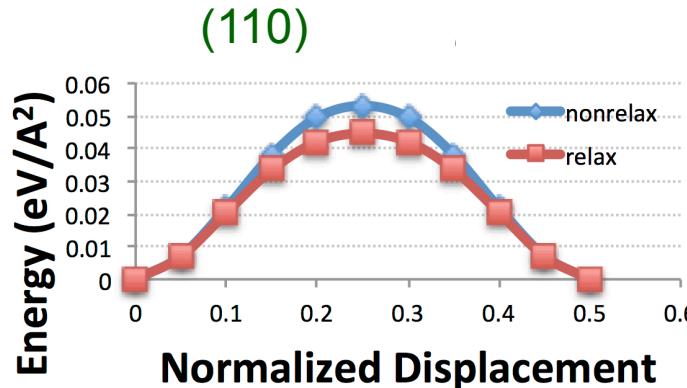


# Ta SNAP potential was fit to a DFT-based training set containing ‘usual suspects’

For each configuration in training set, fit total energy, atomic forces, stress

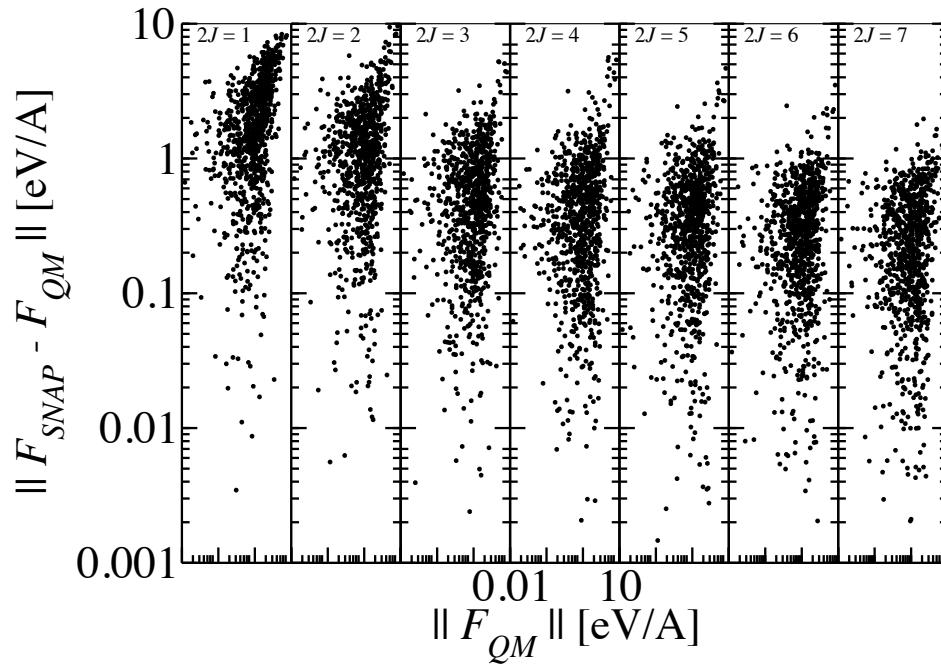
- Equilibrium lattice parameter
- Elastic constants ( $C_{11}$ ,  $C_{12}$ , and  $C_{44}$ ) and bulk modulus (B)
- Free surface energies: (100), (110), (111), and (112)
- Generalized planar stacking fault curves: {112} and {110}
- Energy-Volume (Contraction and Dilation) - BCC, FCC, HCP, and A15
- Lattices with random atomic displacements
- Liquid structure

Example: DFT-based Generalized Stacking Fault Energies



# Effect of Higher-order Bispectrum Components

- Liquid force errors decrease with increasing  $J$
- Diminishing returns beyond  $J = 7/2$

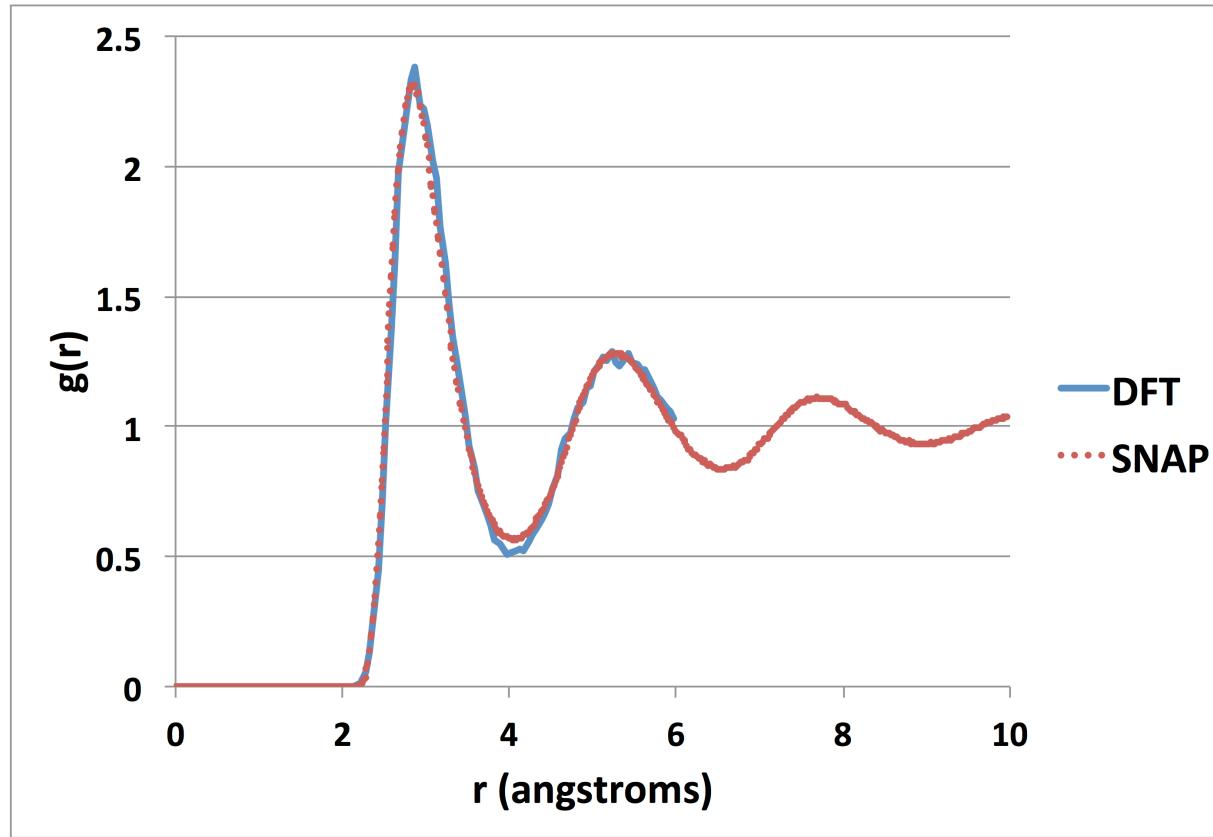


$2J$	N	Ferr
1	2	2.09
2	5	1.39
3	8	0.66
4	14	0.53
5	20	0.44
6	30	0.35
7	40	0.30

# SNAP potential yields good agreement with DFT results for some standard properties

	DFT	SNAP	Zhou (EAM)	ADP
Lattice Constant (Å)	3.320	3.316	3.303	3.305
B (Mbar)	1.954	1.908	1.928	1.971
$C' = (1/2)(C_{11} - C_{12})$ (Mbar)	50.7	59.6	53.3	51.0
$C_{44}$ (Mbar)	75.3	73.4	81.4	84.6
Vacancy Formation Energy (eV)	2.89	2.74	2.97	2.92
(100) Surface Energy (J/m <sup>2</sup> )	2.40	2.68	2.34	2.24
(110) Surface Energy (J/m <sup>2</sup> )	2.25	2.34	1.98	2.13
(111) Surface Energy (J/m <sup>2</sup> )	2.58	2.66	2.56	2.57
(112) Surface Energy (J/m <sup>2</sup> )	2.49	2.60	2.36	2.46
(110) Relaxed Unstable SFE (J/m <sup>2</sup> )	0.72	1.14	0.75	0.58
(112) Relaxed Unstable SFE (J/m <sup>2</sup> )	0.84	1.25	0.87	0.74

# Liquid structure: SNAP and DFT are in excellent agreement

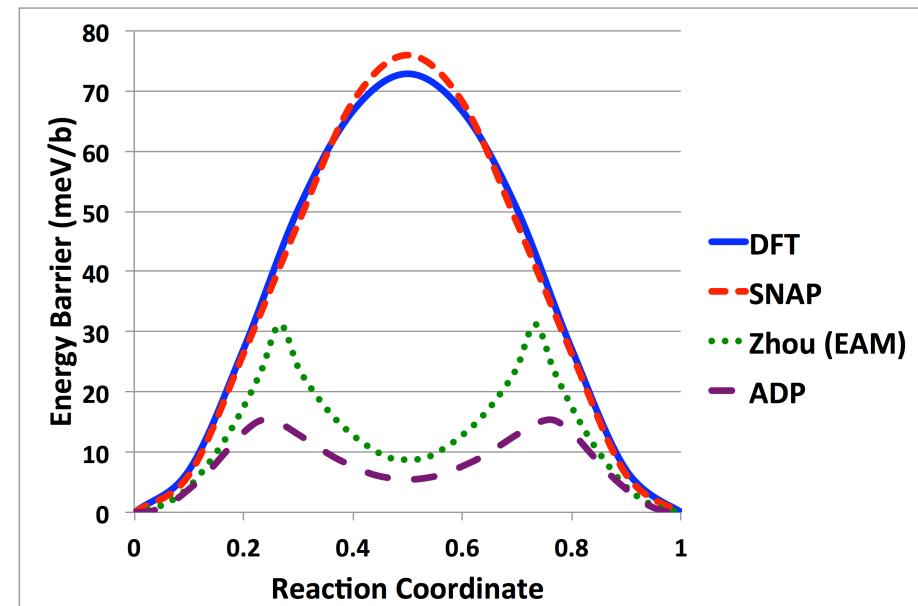
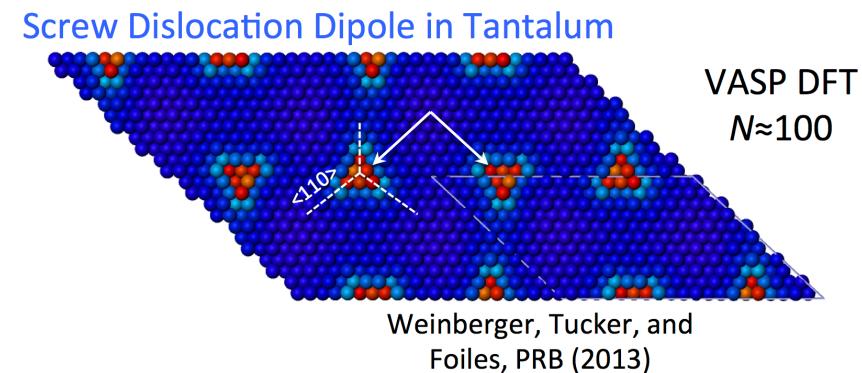


Liquid pair correlation function,  $g(r)$  computed at 3250 K (~melting point) and experimental density

- DFT: 100 atoms, 2 picoseconds
- SNAP: 1024 atoms, 200 picoseconds

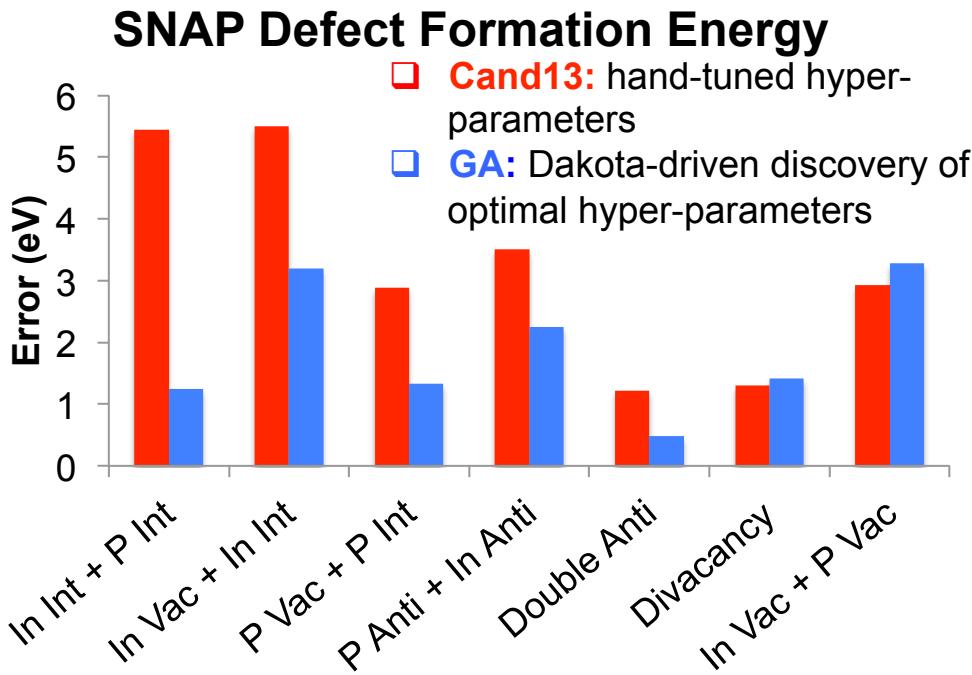
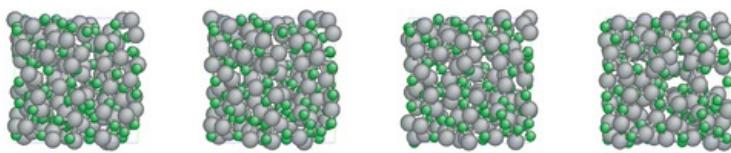
# SNAP potentials predict correct Peierls barrier for Ta screw dislocations

- Peierls barrier is the activation energy to move a screw dislocation
- Many simple interatomic potentials incorrectly predict a metastable state
  - Leads to erroneous dynamics
- SNAP potential agrees well with DFT calculations
  - Future work will explore dislocation dynamics based on this potential



Thompson et al. arxiv.org/abs/1409.3880  
J. Comp. Phys. (2015)

# SNAP Indium Phosphide



## Additional Challenges

- Two elements
- Different atom sizes
- Diverse structures
- Defect formation energies
- Sensitive to curvature

## Innovations

- Differentiate elements by: density weight, linear coefficients, neighbor cutoff
- Trained against relaxed defect structures
- Trained against deformed defect structures

## Result (so far)

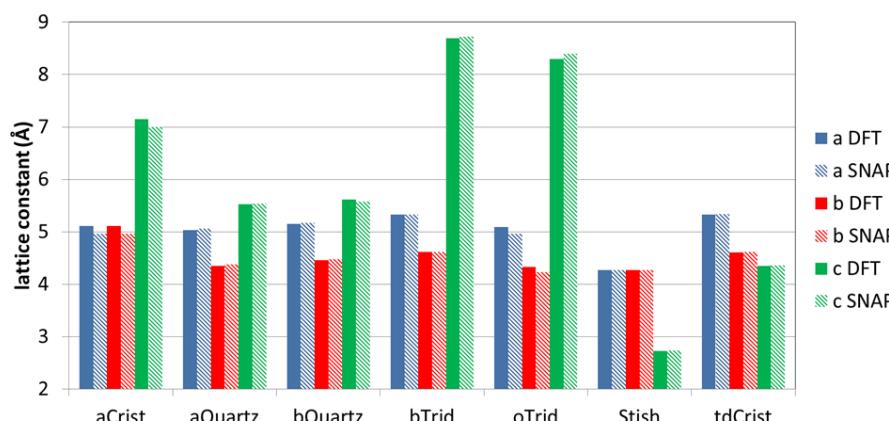
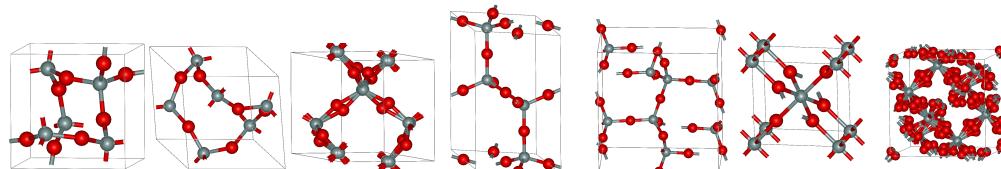
- Good overall fit
- Defect energy error > 1 eV

# SNAP Silica: Promising Start

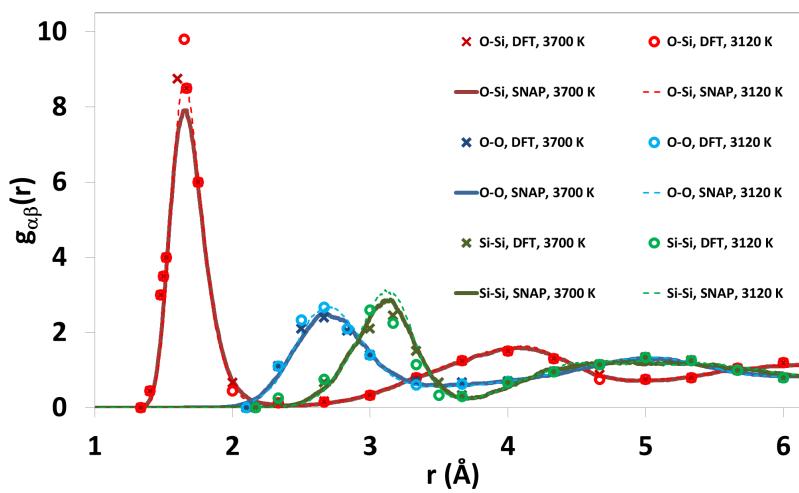
(Stan Moore, Paul Crozier, Peter Schultz)



Less than 3% error in predicted lattice parameters of 7 crystal polymorphs



Good agreement with QM liquid structure for SiO<sub>2</sub>



## Additional Challenges

- Electrostatics
- Started with no training data
- Goal: quantum-accurate prediction of Si/SiO<sub>2</sub> interface

## Innovations

- Generated training data adaptively, on-the-fly
- Added fixed point charges, long-range electrostatics

## Result (so far)

- Good agreement with QM for SiO<sub>2</sub> crystal polymorphs
- Good agreement with QM liquid structure for SiO<sub>2</sub>

# Conclusions

- SNAP is a new formulation for interatomic potentials
  - Geometry described by bispectrum components
  - Energy is a linear regression of bispectrum components
- Works well for Ta
  - Liquid structure
  - Peierls barrier for screw dislocation motion
- Ongoing work
  - Extension to binary systems: InP, SiO<sub>2</sub>, TaO<sub>x</sub>
- SNAP Ta potential published
  - arxiv.org/abs/1409.3880
  - J. Comp. Phys. (2015)
- SNAP Ta available in LAMMPS

## Primary Collaborators

Laura Swiler  
Stephen Foiles  
Garrett Tucker

## Additional Collaborators

Christian Trott  
Peter Schultz  
Paul Crozier  
Stan Moore  
Adam Stephens

# FitSnap.py: Robust Software Framework

Key advantages of fitsnap.py

- Minimal file I/O
- Use of NumPy/SciPy
- Caching and reuse of data
- File-based input
- Supports parallel LAMMPS

## Hyper-parameter Optimization

