

Exceptional service in the national interest

# **QMC** WORKSHOP 2025 WORKFLOW FOR SOLID-STATE CALCULATIONS

**CPSFM** 

Center for Predictive Simulation of Functional Materials











Raymond C. Clay, III

July 15, 2025

CONTROLLED BY:

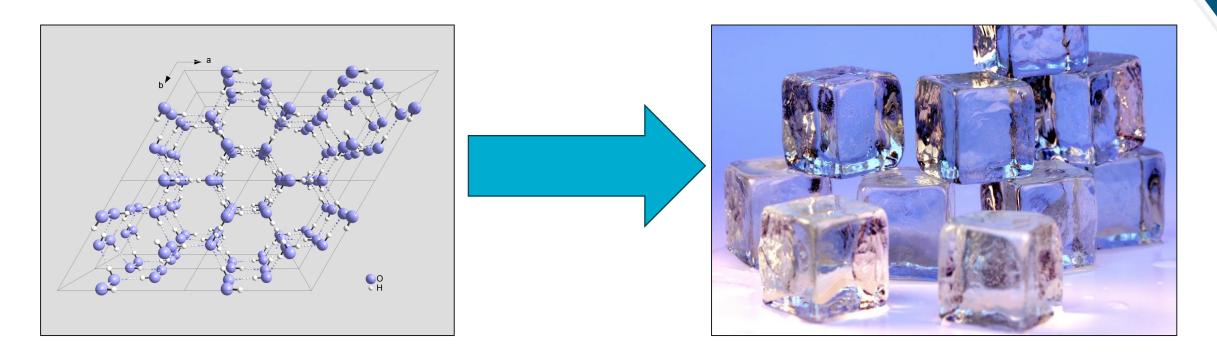




Sandia National Laboratories is a multimission laboratory managed and operated by National Technology and Engineering Solutions of Sandia, LLC, a wholly owned subsidiary of Honeywell International Inc., for the U.S. Department of Energy's National Nuclear Security Administration under contract DE-NA0003525



#### **MODELLING MATERIALS**

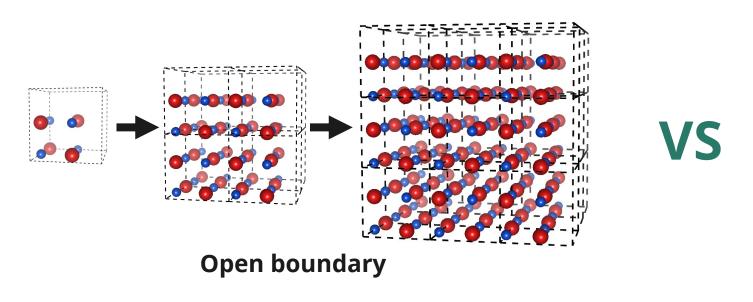


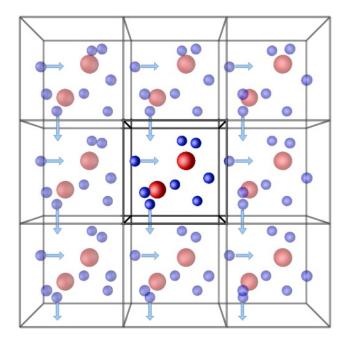
- In contrast to molecules, we are trying to determine properties of macroscopic objects.
- Micron or larger, ~10^23 atoms.
- Consider an idealized scenario, the "thermodynamic limit" ( L=∞, N=∞ )
- The challenge is how to predict these properties in the "thermodynamic limit" with only 10-1000 atoms.

## **OVERVIEW**

- Posing the problem—Boundary Conditions.
- Finite-size effects in DFT.
- One and two-body finite size effects for QMC.
- QMC wave functions for solids.
- Overview of Lab.

#### PERIODIC BOUNDARY CONDITIONS





**Periodic Boundary** 

- For molecules, system is really straightforward to write down.
- If we try to build up a solid state system (left) using the same techniques, we introduce surfaces an surface physics. This is NOT reflective of bulk properties.
- Imagine instead a finite system, but instead of being surrounded by vacuum, we have an infinite lattice of copies of our unit cell.
- No surface. Atoms pass out of the box, they come in from the other side. They "wrap around".
- This is what we mean by "periodic boundary conditions". Converges way faster to TDL.



#### WHAT IS A "FINITE-SIZE EFFECT"?

- For otherwise converged calculations, a "finite-size effect" is any deviation from the infinite system answer (in almost all cases, the "right" answer).
- Examples: energy/cell, pressure, electron density, etc.
- If you plot a quantity vs. system size and see that it changes, it's likely a finite-size effect that you need to control for.

#### DFT AND BLOCH'S THEOREM

**Bloch's theorem**—For electrons in a perfect crystal, there is a basis of wave functions with the following two properties:

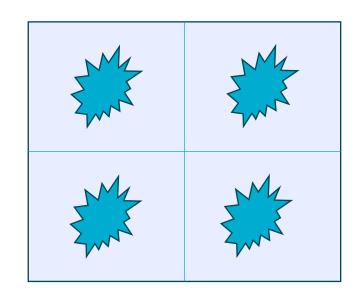
- each of these wave functions is an energy eigenstate,
- each of these wave functions is a Bloch state, meaning that this wave function  $\psi$  can be written in the form  $\psi(\mathbf{r}) = e^{i\mathbf{k}\cdot\mathbf{r}}u(\mathbf{r})$ , where  $u(\mathbf{r})$  has the same periodicity as the atomic structure of the crystal, such that  $u_{\mathbf{k}}(\mathbf{x}) = u_{\mathbf{k}}(\mathbf{x} + \mathbf{n} \cdot \mathbf{a})$ .

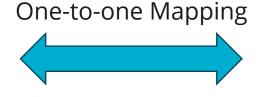
$$\hat{H}_{\mathbf{k}}u_{\mathbf{k}}(\mathbf{r}) = \left[rac{\hbar^2}{2m}(-i
abla+\mathbf{k})^2 + U(\mathbf{r})
ight]u_{\mathbf{k}}(\mathbf{r}) = arepsilon_{\mathbf{k}}u_{\mathbf{k}}(\mathbf{r})$$

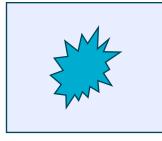
- Kohn-Sham DFT is a single-particle theory, so Bloch's theorem holds.
- Cool, but what does this mean in practice?
- \*\*STRONGLY RECOMMEND READING ASHCROFT & MERMIN OR RICHARD MARTIN'S "ELECTRONIC STRUCTURE THEORY" TO LOCK THIS DOWN IN YOUR BRAIN\*\*\*

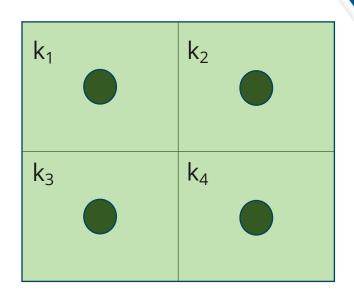
# **BLOCH'S THEOREM**









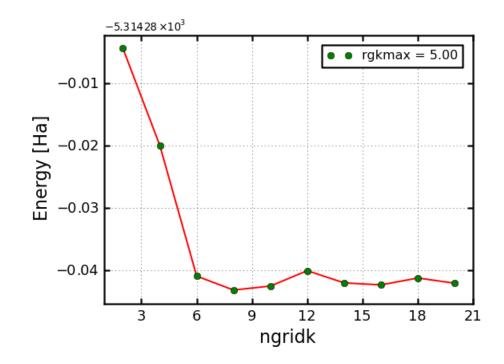


$$\hat{H}\psi(\mathbf{r}) = igg(-rac{\hbar^2}{2m}
abla^2 + \mathit{U}(\mathbf{r})igg)\psi(\mathbf{r}) = \epsilon\psi(\mathbf{r})$$

$$\hat{H}_{\mathbf{k}}u_{\mathbf{k}}(\mathbf{r}) = \left[rac{\hbar^2}{2m}(-i
abla + \mathbf{k})^2 + U(\mathbf{r})
ight]u_{\mathbf{k}}(\mathbf{r}) = arepsilon_{\mathbf{k}}u_{\mathbf{k}}(\mathbf{r})$$

- One-to-one mapping between the two pictures.
- Instead of solving Schrodinger's equation for the whole cell (left), I can solve it the primitive cell at over a set of k-points.
- Significantly cheaper, because we work with way fewer orbitals in the primitive cell.
- For semi-local functionals (LDA, PBE), there is no coupling between k-points, and so I can do these calculations in parallel.

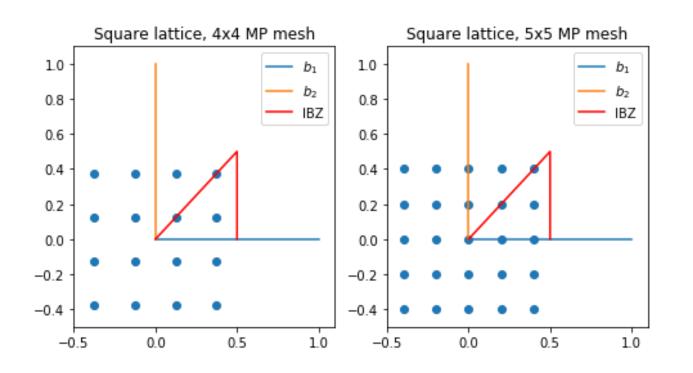
# FINITE SIZE EFFECTS IN DFT (ONE-BODY)



http://exciting.wikidot.com/lithium-simple-convergence-tests

- (Left) Convergence study of total energy w.r.t. kpoint grid size (ngridk x ngridk x ngridk) for lithium.
- Once the density is converged (SCF step), there are many quantities that are expressible as  $I = \int d^3 \mathbf{k} \ f(\mathbf{k})$ . The finite-size error here is an integration error caused by using a finite resolution grid. This can be fixed by using larger (finer) k-point grids.
- In effect, the standard SCF and NSCF convergence procedures you are already accustomed to completely eliminate electronic finite-size effects in DFT.

#### FINAL NOTE ON BRILLOUIN ZONE INTEGRATION



https://sites.psu.edu/dftap/ 2019/02/01/shifting-ofkpoints-in-hexagonallattices/

- We can use space group symmetries to cut down the number of k-points we need to consider.
- The region outlined in red is the "irreducible Brillouin zone". By symmetry, these are the only "unique" or irreducible k-points we need to describe to evaluate any integral over the entire Brillouin zone.
- If enabled, DFT codes will list these inequivalent k-points with "weights", basically how many other k-points are related to it by symmetry. More on this coming soon.

## ONE-BODY FINITE SIZE EFFECTS IN QMC

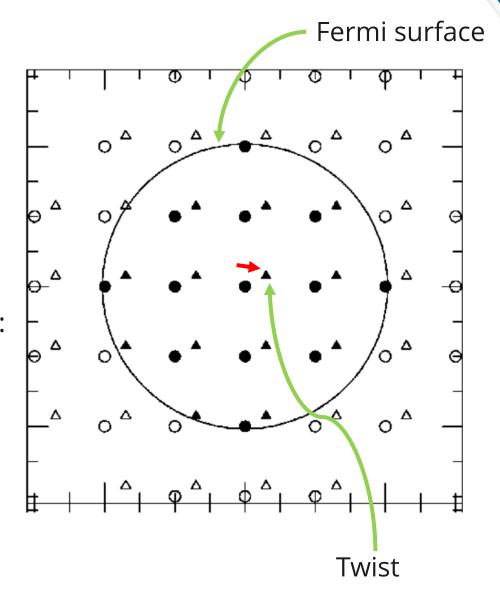
- Many-body wave functions spoil the power of Bloch's theorem, but we know that there are sizeable one-body finite size effects present, just from the mean field level.
- We will use a wave function ansatz as follows:

$$\Psi(\mathbf{r}_1, \mathbf{r}_2 + \mathbf{L}, \dots, \mathbf{r}_N) = e^{i\mathbf{L}\cdot\mathbf{k}}\Psi(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N)$$

We define "twist-averaged" observables as follows:

$$\langle \hat{A} 
angle_{TABC} = rac{\int d\mathbf{k} \int d\mathbf{R} \langle \Psi(\mathbf{R},\mathbf{k}) | \hat{A} | \Psi(\mathbf{R},\mathbf{k}) 
angle}{\int d\mathbf{k} \int d\mathbf{R} \langle \Psi(\mathbf{R},\mathbf{k}) | \Psi(\mathbf{R},\mathbf{k}) 
angle}$$

- Reduces finite size effects<sup>1</sup> (more on this later)
- Faster convergence to thermodynamic limit<sup>1</sup>



# (CANONICAL) TWIST AVERAGING PROCEDURE

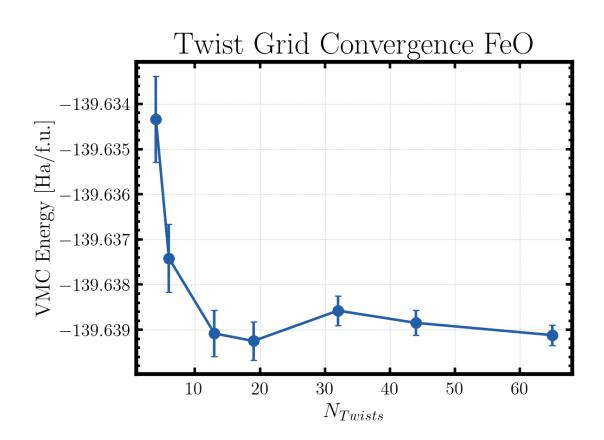
- Must apply TABC to all observables, not just energy!
- Note that the integral over the VMC/DMC random walk and twist vector is an average, and so can be discretely calculated like one.

$$\langle \hat{A} 
angle_{TABC} = \sum_{\mathbf{k}} A(\mathbf{k}) w_{\mathbf{k}}$$

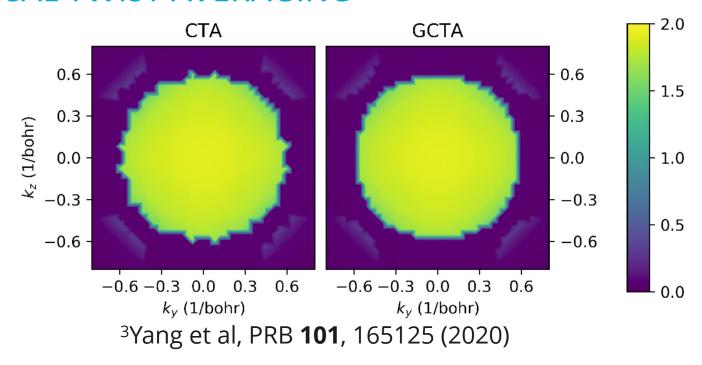
- A(k) is the usual MC average of A for a particular twist
- $w_k$  is the corresponding weight (just like a k-point mesh in DFT). Above expression assumes sum of weights is 1. Alternatively, the weight is the multiplicity of that k-point divided by the total number of kpoints.

#### TWIST-AVERAGED BOUNDARY CONDITIONS IN PRACTICE

- Good news! Twists are independent. (Like in DFT)
- Error reduction  $\propto \frac{1}{\sqrt{N_{Twists}}}$
- How many twists are enough?
  - Depends on size, metallic vs insulating
  - You must check this yourself!
- Nexus can generate primitive cell k-point to supercell twist grid mapping for you



#### **GRAND-CANONICAL TWIST AVERAGING**

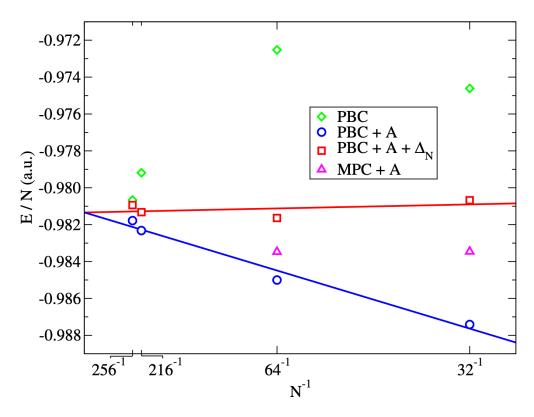


- It's possible to do better than canonical twist averaging, both in terms of convergence rate w.r.t. twists, and with respect to biases.
- Canonical twist averaging occupies each with the actual number of electrons in the cell.
- Grand-canonical twist averaging sets the occupation based on the Fermi level.
- Does less violence to the Fermi surface. Try it out, find it in the manual.
- A. Annaberdiyeev, P. Ganesh, J. T. Krogel J. Chem. Theory Comput. 2024, 20, 2786–2797

# Th

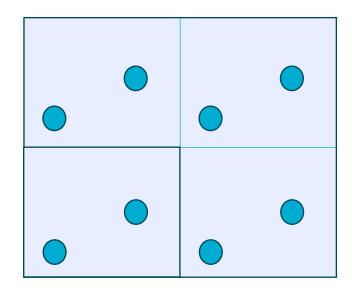
#### ARE WE DONE?

- Unfortunately, no. This procedure mitigates single-particle finite size effects, but QMC is a many-body theory. So we have many-body finite size effects.
- Fortunately, the only ones we have to worry about for energetics are two-body finite size effects.
- Analytic results in 3D and ample empirical evidence confirms that the two-body finite size effect in bulk systems scales like 1/N in the number of atoms.
- When in doubt, run several (twist averaged) supercells, fit to 1/N, and extrapolate to 1/N=0 (N=infinity).

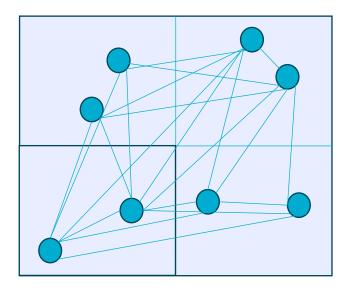


Diamond silicon energy/atom as a function of 1/N. PBC+A is "twist averaging". Difference between N=32 and N=infinity is ~6mHa/atom, which is large.

#### TWO-BODY FINITE SIZE EFFECTS



Bottom Left Cell in PBC's



Bottom Left Cell in 2x2 Supercell

- Electrons interact via a long range coulomb potential. This induces correlations between cells that cannot be captured in a Bloch picture. That is, they will redistribute according to the positions of other electrons in the supercell.
- Won't show up in density, but WILL show up in two-body correlation functions (g(r), S(k)). These impact the energy significantly (as seen previously).

#### HOW DO WE FIX TWO-BODY FINITE SIZE EFFECTS?

- 1. Extrapolate
  - 1. Already talked about.
  - 2. Gold standard, and easily validated. Will get really expensive.
- 2. Correct
  - 1. MPC (model periodic coulomb potential)
  - 2. Structure factor and Jastrow based corrections (Chiesa corrections)
  - 3. KZK (DFT Based)

#### MODEL PERIODIC COULOMB POTENTIAL

- Ewald gets Hartree energy exact, but biases XC energy
- Model Periodic Coulomb (MPC) fixes that¹:

$$V_{MPC} = \frac{1}{2} \sum_{i \neq j} f(r_i - r_j) + \sum_{i} \int_{\Omega} \rho(r) [V_e(r_i - r) - f(r_i - r)] dr - \frac{1}{2} \int_{\Omega} \rho(r) \rho(r') [V_e(r - r') - f(r - r')] dr dr'$$
Normal Coulomb inside Modified potential outside

- Not a real FSC by itself, but sometimes used in that way
- See MPC potential for details

# STRUCTURE FACTOR (CHIESA) CORRECTIONS

- Gold standard: Use structure factor<sup>1,2,3</sup>
- PE per particle is related to s(k) via:

$$V_N = \frac{2\pi}{\Omega} \sum_{\mathbf{k} \neq 0} \frac{1}{k^2} \left( s(\mathbf{k}) - 1 \right)$$

Correction is an integration error:

$$\Delta V_N = \frac{1}{4\pi^2} \int d\mathbf{k} \, \frac{s(\mathbf{k}) - 1}{k^2} - \frac{2\pi}{\Omega} \sum_{\mathbf{k} \neq 0} \frac{s_N(\mathbf{k}) - 1}{k^2}$$

Use SkAll + qmcfinitesize in postprocessing

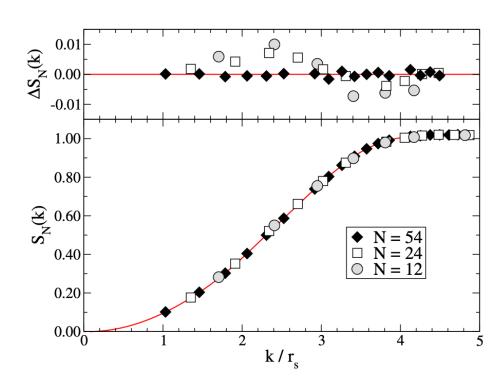


FIG. 1 (color online). Lower panel: static structure factor for the electron gas at  $r_s = 10$ . Upper panel:  $\Delta S_N = S_N(k) - S_{66}(k)$ . The difference is computed using a spline function interpolation of  $S_{66}$ .

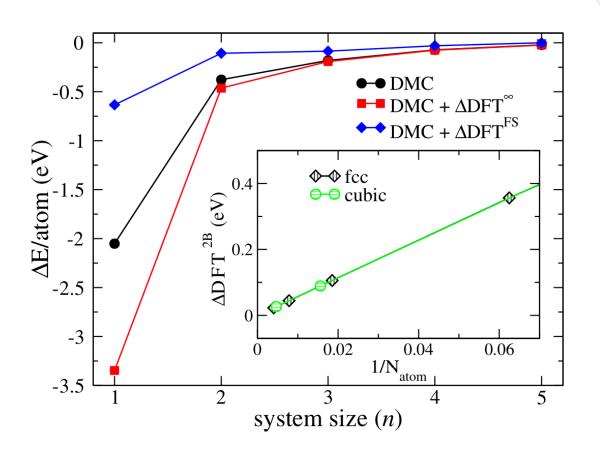
<sup>&</sup>lt;sup>1</sup>Chiesa et al. PRL **97**, 076404 (2006)

<sup>&</sup>lt;sup>2</sup>Drummond et al. *PRB* **78**, 125106 (2008)

<sup>&</sup>lt;sup>3</sup>Holzmann et al. PRB **94**, 035126 (2016)

#### **KZK CORRECTION**

- KZK correction<sup>1</sup>:
- Take  $E_{XC}(N)$  from the HEG
- In ESPRESSO: dft="kzk"
  - LDA only by default
- Entirely independent of QMC
- Gives approximate 2-body KE & PE corrections



<sup>&</sup>lt;sup>1</sup>Kwee et al. PRL **100**, (2008)

UUR

# WAVE FUNCTIONS FOR SOLIDS

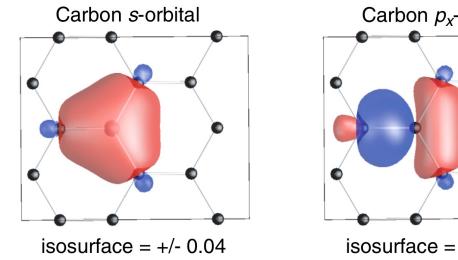
## **QMC WAVE FUNCTIONS**

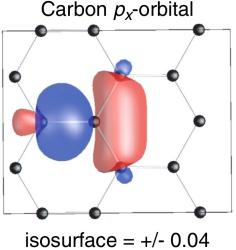
$$\Psi^{SJ}(X) = e^{J(X)} \det \begin{vmatrix} \phi_1(\mathbf{x}_1) & \phi_1(\mathbf{x}_2) & \cdots & \phi_1(\mathbf{x}_N) \\ \phi_2(\mathbf{x}_1) & \phi_2(\mathbf{x}_2) & \cdots & \phi_2(\mathbf{x}_N) \\ \vdots & \vdots & \ddots & \vdots \\ \phi_N(\mathbf{x}_1) & \phi_N(\mathbf{x}_2) & \cdots & \phi_N(\mathbf{x}_N) \end{vmatrix}$$

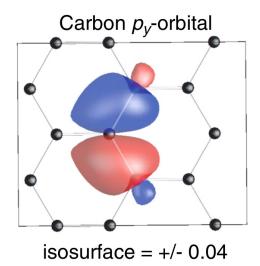
- At a very high level, this is the same ansatz and general workflow as for molecules, but solids
  pose interesting additional complications.
- Plane wave DFT codes are dominant in this space for many reasons.
- Systematically convergeable as we increase plane wave cutoff, but the memory requirements can become unwieldy as we increase system size.
- Heavy elements can also significantly increase plane wave cutoffs and thus overall cost of simulation.
- Speed is important too! Our team hits 1000's of electrons more frequently in solids than in molecules.

## **QMC WAVE FUNCTIONS FOR SOLIDS -- ORBITALS**

Key idea: Use 3d b-splines to represent each orbital<sup>1</sup>





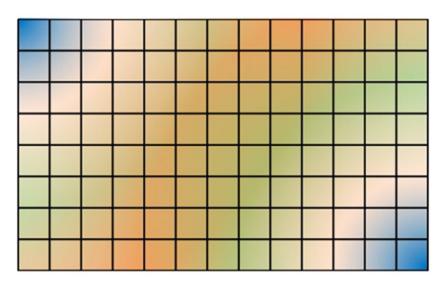


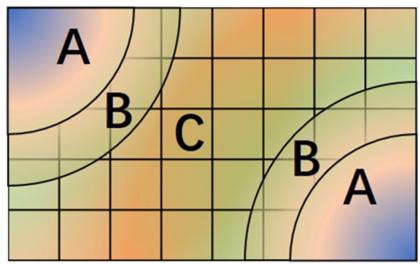
- Evaluate 64 grid points per orbital independent of system and basis set size
- Efficient routines transform DFT orbitals to splines automatically
- These get BIG. Using shared memory and proper batching is pretty much required for production calculations.

<sup>&</sup>lt;sup>1</sup>Alfe and Gillan, PRB **70**, 161101 (2004) Fig from: Marzari et al. RMP 84, (2012)

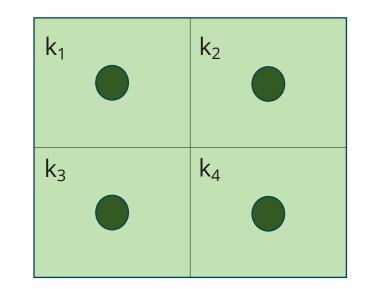
## QMC WAVE FUNCTIONS FOR SOLIDS – HYBRID REPRESENTATION

- We can further reduce memory costs using a hybrid representation<sup>1</sup>
- Use spherical harmonics near nucleus (A)
- Smooth transition region (B)
- Normal 3d B-splines in intersitial space (C)
- Memory reduction of 4-8x possible!
- Controlled by hybridrep tag

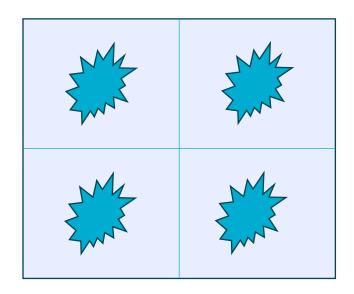




#### WAVE FUNCTION GENERATION





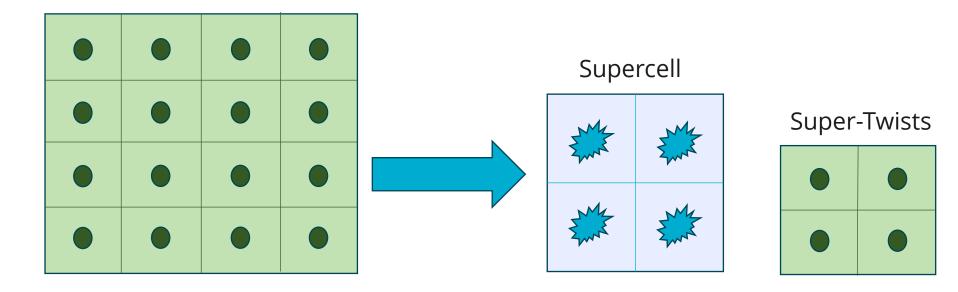


$$\hat{H}_{\mathbf{k}}u_{\mathbf{k}}(\mathbf{r})=\left[rac{\hbar^{2}}{2m}(-i
abla+\mathbf{k})^{2}+U(\mathbf{r})
ight]u_{\mathbf{k}}(\mathbf{r})=arepsilon_{\mathbf{k}}u_{\mathbf{k}}(\mathbf{r})$$

$$\hat{H}\psi(\mathbf{r}) = igg(-rac{\hbar^2}{2m}
abla^2 + \mathit{U}(\mathbf{r})igg)\psi(\mathbf{r}) = \epsilon\psi(\mathbf{r})$$

- We typically use Bloch's theorem when possible to accelerate generation of single particle orbitals for use in the QMC supercell.
- We will run primitive (or small) cell calculations with a large number of k-points, and then "unfold" those by Bloch's theorem into the super cell.

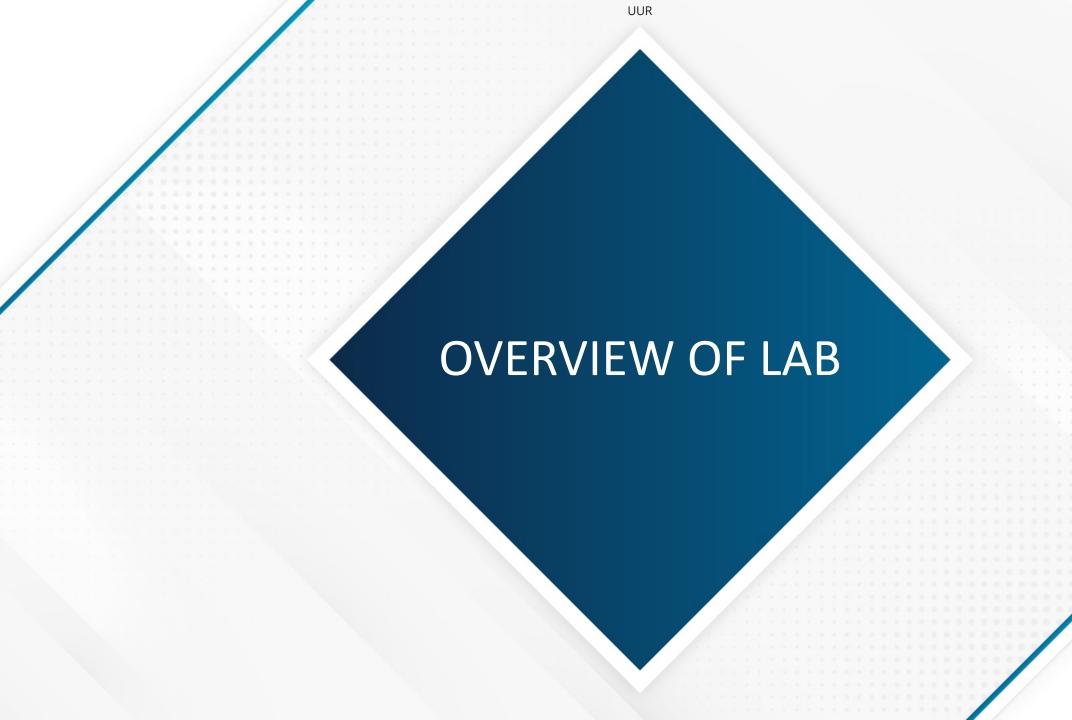
## WAVE FUNCTION GENERATION—MORE REALISTIC USE CASE



- Remember, we need to do twist-averaging to take care of one-body finite size effects, but we
  also need to run multiple supercells to be able to extrapolate or correct the two-body finite
  size effects.
- Proper generation of primitive cell twists allows us to generate orbitals for a supercell with twists.
- Nexus is smart enough to do this, and you will do this in the lab.

## CHECKLIST FOR QMC ON SOLIDS

- 1. Converge twist grid
- 2. Exploit symmetry
- 3. Plan ahead for estimators/observables
- 4. Establish work required for desired accuracy in a small cell
- 5. Plan to quantify/mitigate finite size effects
- 6. When in doubt, get a bigger supercell!
- 7. Reach out to QMCPACK community: <a href="mailto:qmcpack.slack.com">qmcpack.slack.com</a>



#### LAB SUMMARY: DIAMOND

- 1. Generate converged scf density.
- 2. Use density to generate SPO's for 8 atom cell with NSCF procedure.
- Optimize J1 and J2.
- 4. Perform VMC and DMC calculations for (1x1x1), (2x2x2), and (3x3x3) twist grids.
- Twist-average the results of each grid calculation to determine the twist-averaged total energy for each grid.
- 6. Convergence figure for total energy including (4x4x4) and (6x6x6) grids.

