# Melt ascent and differentiation from the upper mantle to the lower crust

Tobias Keller

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## 1 Introduction

In continental arc and rift volcanism erupting lavas are thought to be fed through trans-crustal magma processing systems extending down into the deep crust and mantle lithosphere. The magma source, however, is located even deeper down in the mantle asthenosphere, where decompression by upward mantle flow and/or fluxing by aqueous fluid infiltration lead to partial melting of hot mantle rock. The processes by which mantle melts first ascend into the base of the continental plate above and their rates and spatial distribution remain poorly understood. Current gaps in knowledge are due to, on one hand, the challenges in gaining sufficiently resolved data on these regions by geophysical imaging methods. The materials involved can only be sampled in few locations where lower crustal section have been tilted and eroded to reveal the roots of a transcrustal magmatic system, or through occasional xenoliths, rock pieces brought from depth to the surface in fast ascending magmas. On the other hand, modelling 12 of melt ascent into the base of a plate is challenging due to a range of complex nonlinearities 13 inherent in the problem. Mechanical interactions between melt and host rock are prone to instabilities and localisation feedbacks, leading to narrowly localised zones of enhanced melt flow 15 and rock deformation. Thermochemical reactions between the phases can also lead to localising feedbacks such as dissolution channels solidification chimneys. 17

One possible way to gain better understanding of melt ascending into the deep crust is to use the chemistry of plutonic and volcanic igneous rocks as records of the dynamic process environments experienced during magma ascent and emplacement. This task is complicated by the fact that geochemical signatures in final rock products are the result of reaction and transport rates acting along a potentially protracted and circuitous trajectory through pressure-

temperature-composition space. Interpreting final compositions in terms of mass balancing models, whereby a composition is modelled as a mixture of assumed end-members (e.g., source rock, fractionated crystals, crustal assimilant), is widely practised but results are inherently non-unique.

In this contribution, we present a coupled thermo-chemical-mechanical model of melt ascent 27 into the base of an overriding plate. The model includes two-phase mechanics including non-28 Newtonian rheology and decompaction failure, as well as a thermo-chemical model considering 29 the evolution of melt fraction as a function of temperature and major-element composition, as 30 well as tracking compactible and incompatible trace elements, and radiogenic and stable isotopic composition. The geochemical evolution model employs an idealised approach whereby the 32 complex geochemistry involving a dozen major elements, a few minor elements and volatiles, 33 tens of trace elements, and a diversity of radiogenic decay and stable isotope systems are reduced to few idealised components approximating some of the key behaviours of each aspect 35 of geochemistry. 36

## $_{\scriptscriptstyle 7}$ 2 Method

#### $^{_{18}}$ $\;2.1$ Mechanical model

Melt ascent is modelled as the porous flow of an incompressible, viscous fluid through a permeable and deformable matrix of incompressible, visco-plastic solid subject to decompaction failure. The non-dimensionalised governing equations expressing conservation of mass and momentum in the two-phase mixture are,

$$\nabla P = -\nabla p + \nabla \cdot \boldsymbol{\tau} + \phi B \hat{\mathbf{z}} , \qquad (1a)$$

$$\nabla \cdot \mathbf{V} = -\nabla \cdot \mathbf{v} , \qquad (1b)$$

where  $\mathbf{V} = [U, V, W]$  is the dynamic velocity, and P the dynamic pressure of the mixture,  $\mathbf{v} = [u, v, w]$  the segregation velocity of the pore fluid, and p the compaction pressure and the shear stress tensor of the solid matrix.  $\nabla = [\partial/\partial x, \partial/\partial y, \partial/\partial z]$  is the partial spatial derivative. All variables and parameters are functions of position,  $\mathbf{x} = [x, y, z]$ , and time, t, unless indicated otherwise by a subscript nought. The unit vector of the depth coordinate,  $\hat{\mathbf{z}}$ , points in the direction of gravity. The dimensionless number B expresses the magnitude of buoyancy forces to tensile strength of the aggregate (more details below).

The dynamic velocity and pressure are each reduced by an invariant background field with constant spatial gradients. The dynamic pressure, P, is the Stokes pressure of the mixture reduced by a lithostatic background field,  $P_b$ , with  $\nabla P_{\text{lith}} = [0, \rho_0 g_0]$ , where  $\rho_0$  is the constant solid matrix density, and  $g_0$  the gravity constant. The dynamic velocity is the Stokes velocity of the mixture reduced by a background flow field,  $V_b$ , which can assume both pure- and simple-shear flow patterns adjusted by dimensionless factors Pu, and Si which control the magnitude of shear stress induced by backround deformation relative to the tensile strength of the aggregate.

The non-dimensionalised closures for segregation velocity,  $\mathbf{v}$ , compaction pressure, p, shearstress tensor,  $\underline{\boldsymbol{\tau}}$ , and visco-plastic matrix rheology are,

$$\mathbf{v} = -K\left(\mathbf{\nabla}P + \mathbf{B}\hat{\mathbf{z}}\right) , \qquad (2a)$$

$$p = -\frac{\eta}{\phi} \nabla \cdot \mathbf{V} , \qquad (2b)$$

$$\underline{\boldsymbol{\tau}} = \eta \, \underline{\dot{\boldsymbol{\varepsilon}}} \,\,, \tag{2c}$$

$$K = \left(\frac{\phi}{\phi_0}\right)^m \exp\left(-E^{\ell} \left[\frac{1}{T} - \frac{1}{T_0}\right]\right) , \qquad (2d)$$

$$\eta = \min \left( \exp \left( E^s \left[ \frac{1}{T} - \frac{1}{T_0} \right] - \frac{\lambda}{F} (\phi - \phi_0) \right) \left( \frac{\dot{\varepsilon}}{\dot{\varepsilon}_b} \right)^{-n}, \quad \frac{\tau_y}{\dot{\varepsilon}} + \eta_{\min} \right), \tag{2e}$$

$$\tau_y = 1 + p$$
 (2f)

 $\hat{\mathbf{z}}$  is the unit vector pointing along the depth coordinate. The shear strain rate tensor,  $\dot{\underline{\boldsymbol{\varepsilon}}}$ , its components, and magnitude are defined as,

$$\underline{\dot{\boldsymbol{\varepsilon}}} = \frac{1}{2} \left( \boldsymbol{\nabla} \mathbf{V} + \boldsymbol{\nabla} \mathbf{V}^T \right) - \frac{1}{3} \boldsymbol{\nabla} \cdot \mathbf{V} \underline{\mathbf{I}} , \qquad (3a)$$

$$\dot{\varepsilon} = \sqrt{\frac{1}{2}\underline{\dot{\varepsilon}} : \underline{\dot{\varepsilon}}} . \tag{3b}$$

The rheological coefficients are centred on the reference melt fraction,  $\phi_0$ , temperature,  $T_0$ , and strain-rate  $\dot{\varepsilon}_b$ . The matrix creep viscosity is weakened exponentially by the presence of melt with a factor  $\lambda = 30$ , by inverse temperature with the solid activation energy  $E^s$ , and

is softening as a powerlaw of the strain rate magnitude,  $\dot{\varepsilon}$ , with a non-Newtonian powerlaw exponent n=2/3. The reference strain-rate for the non-Newtonian powerlaw is determined by the applied pure- and/or simple-shear background deformation,  $\dot{\varepsilon}_b = |\text{Pu}| + |\text{Si}|$ . Hence, the dimensionless shear viscosity,  $\eta$ , at  $\phi = \phi_0$ ,  $T = T_0$ , and  $\dot{\varepsilon} = \dot{\varepsilon}_b$  is equal to unity, and the compaction viscosity equal to  $\phi^{-1}$ . Decompaction failure is implemented using an effective-viscosity approach limiting shear stress as a function of compaction pressure so as not to exceed the tensile failure criterion,  $\tau_y$  (see Fig. ??). A minimum yield viscosity,  $\eta_{\text{min}}$ , is added to regularise the problem by imposing a minimum length scale of localised failure zones.

To highlight the role of decompaction failure, we have scaled pressures and stresses by the 85 tensile strength of the matrix,  $p_o = \sigma_{T,0}$  to non-dimensionalise the problem. The dimensionless 86 number B =  $\Delta p_0/\sigma_{T,0}$  expresses the relative magnitude of the buoyancy-related phase pressure 87 difference,  $\Delta p_0 = \Delta \rho_0 g_0 \delta_0$ , relative to the tensile strength scale. The former is a natural scale 88 of the Stokes-Darcy problem proportional to the matrix-melt density difference,  $\Delta \rho_0$ . Velocities 89 are scaled by the Darcy speed scale  $w_0 = K_0 p_0 / \ell_0$ , and length by  $\ell_0 = \sqrt{\eta_0 K_0}$ , a natural scale 90 of the problem related to the segregation-compaction length,  $\delta_0 = \sqrt{\phi_0^{-1}} \ell_0$ , where  $\eta_0$  is the 91 characteristic matrix shear viscosity, and  $K_0 = a_0^2 \phi_0^m/(b_0 \mu_0)$  the Darcy coefficient scale, with  $a_0$ 92 the characteristic matrix grain size,  $b_0 = 100$  a geometrical constant, and  $\mu_0$  the characteristic 93 melt viscosity.

### 95 2.2 Thermo-chemical model

The thermo-chemical evolution by coupled reaction-transport processes in the partially molten mixture is modelled by a set of equations conserving energy, phase, and component mass, and evolving trace and isotopic composition,

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$$\frac{\partial \phi}{\partial t} = \mathbf{\nabla} \cdot (1 - \phi) \mathbf{v}^s + \Gamma , \qquad (4a)$$

$$\frac{\partial T}{\partial t} = -\bar{\mathbf{v}} \cdot \nabla T + \frac{\nabla^2 T}{\text{Pe}_T} - \frac{\Gamma}{\text{St}} , \qquad (4b)$$

$$\frac{\partial \bar{c}_{\text{maj}}}{\partial t} = -\nabla \cdot \overline{c_{\text{maj}}} \nabla + \frac{\nabla^2 \bar{c}_{\text{maj}}}{\text{Pe}_c} , \qquad (4c)$$

$$\frac{\partial \bar{c}_{\text{tra}}}{\partial t} = -\nabla \cdot \overline{c}_{\text{tra}} \nabla + \frac{\nabla^2 \bar{c}_{\text{tra}}}{\text{Pe}_c} , \qquad (4d)$$

$$\frac{\partial \bar{c}_{\text{irp}}}{\partial t} = -\nabla \cdot \overline{c}_{\text{irp}} \nabla + \frac{\nabla^2 \bar{c}_{\text{irp}}}{\text{Pe}_c} - \Gamma_D , \qquad (4e)$$

$$\frac{\partial \bar{c}_{ird}}{\partial t} = -\nabla \cdot \overline{c}_{ird} \nabla + \frac{\nabla^2 \bar{c}_{ird}}{Pe_c} + \Gamma_D , \qquad (4f)$$

$$\frac{\partial c_{\text{ist}}^s}{\partial t} = -\mathbf{v}^s \cdot \nabla c_{\text{ist}}^s + \frac{\nabla^2 c_{\text{ist}}^s}{\text{Pe}_c} - (c_{\text{ist}}^{\Gamma} - c_{\text{ist}}^s) \frac{\Gamma}{1 - \phi} , \qquad (4g)$$

$$\frac{\partial c_{\text{ist}}^{\ell}}{\partial t} = -\mathbf{v}^{\ell} \cdot \nabla c_{\text{ist}}^{\ell} + \frac{\nabla^{2} c_{\text{ist}}^{\ell}}{\text{Pe}_{c}} + (c_{\text{ist}}^{\Gamma} - c_{\text{ist}}^{\ell}) \frac{\Gamma}{\phi} . \tag{4h}$$

The melt fraction,  $\phi$ , evolves as a consequence of compaction, that is, divergence in the 108 solid velocity field,  $\mathbf{v}_s$ , and (volumetric) melting rate,  $\Gamma$ . Sensible heat characterised by the 109 temperature relative to the interval from lowest to highest melting point evolves as a function of 110 advection by the mixture velocity field,  $\bar{\mathbf{v}} = \phi \mathbf{v}^{\ell} + (1 - \phi) \mathbf{v}^{s}$ , diffusion governed by the inverse of 111 the thermal Peclet number,  $Pe_T$ , and sensible to latent heat conversion upon melting governed by the Stefan number, St. The major element composition of the mixture,  $c_{\text{maj}}^- = \phi c_{\text{maj}}^{\ell} +$ 113  $(1-\phi)c_{\mathrm{maj}}^{s}$  evolves by divergence in the chemical flux,  $c_{\mathrm{maj}}^{s}\mathbf{v}^{s}+\phi c_{\mathrm{maj}}^{\ell}\mathbf{v}^{\ell}$ , and chemical diffusion 114 governed by the chemical Peclet number, Pec. Note that chemical diffusivity of major elements 115 is very slow on crustal length scale, hence chemical diffusion could be safely neglected in these 116 models. We include it here for the purpose of regularisation to avoid growth of overly sharp 117 compositional boundaries that can become numerically problematic for advection schemes. The 118 phase compositions,  $c_{\text{maj}}^{\ell,s}$  are functions of temperature and mixture composition defined by a 119 phase diagram. Total mass conservation is observed by only solving for one phase, assuming 120 that the mixture is saturated and both phase fractions sum to unity, and only solving for one 121 major element component assuming that the two components also sum to unity (i.e., the lever 122 rule applies). 123

The geochemical evolution of trace and isotopic compositions does not need to observe total

mass conservation since the elements involved are assumed to be present in trace quantities only. Hence, equations for all components are solved and no unity sum constraints apply. The mixture 126 trace composition,  $c_{\text{tra}}^{-}$  follows the same form of conservation law as major element composition, with phase trace element compositions,  $c_{\rm tra}^{\ell,s}$ , are set by partition coefficients,  $K_{\rm tra}=c_{\rm tra}^s/c_{\rm tra}^\ell$ 128 here held constant for simplicity. Mixture concentrations of a set of parent and daughter isotopes linked by a radioactive decay system follow the same form of conservation law as major 130 and trace components but are linked by the decay rate,  $\Gamma_D = \bar{c}_{\rm irp}/(\ln(2)D_{1/2})$ , proportional 131 to the concentration of the parent isotope,  $\bar{c}_{irp}$ , and the decay number,  $D_{1/2}$ , which is the 132 dimensionless half-life. Stable isotopic composition of the melt and matrix phases are assumed to evolve only by advection on the phase velocity field and mass transfer between phases by 134 phase-change reaction. Diffusion terms are added for the purpose of regularisation.

## 136 2.3 Numerical Implementation

The model is implemented in two dimension on a square domain of  $L \times L$  non-dimensional size.

The initial condition on temperature is a halfspace cooling solution representing a cool plate over
a near-isothermal mantle. The initial major element concentration is a smoothed random noise
to introduce some heterogeneity. The initial melt fraction is set to the equilibrium value given
the temperature and bulk composition as specified by an idealised peritectic-eutectic phase
diagram. The boundary conditions are free slip and insulating on the sides and free in/outflow
and isothermal on the top and base. The side boundaries can be switched to periodic in models
where horizontal simple shear dominates the background velocity.

The numerical implementation discretises the governing equations on a regular square grid of 145  $500 \times 500$  finite-volume cells using a centred staggered-grid finite-difference scheme. Advection 146 terms are implemented using flux-conservative centred or upwind-biased Fromm methods. The 147 equations are solved using a damped Richardson iterative scheme. For reasons of numerical 148 stability the shear visco-plasticity,  $\eta$ , and the phase reaction rate,  $\Gamma$ , are regularised by applying 149 a small amount of Laplacian smoothing and lagged iteratively to avoid numerical oscillations. 150 The mechanical part of the numerical algorithm was benchmarked against a manufactured 151 analytical solution and shows second order error convergence with increasing spatial resolution. 152 The simulation code was developed and tested in Matlab version R2020a and is openly available

 $_{154}$   $\,$  in on github.com/kellertobs/ivrea.