



# Effect of annealing treatment on properties of Ce-doped indium oxide (ICO) transparent conductive oxide films

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## ABSTRACT

Ce-doped indium oxide films were prepared by reactive plasma deposition, and the effects of annealing temperature and annealing time on the electrical properties, optical properties, and film structure of the ICO films were investigated. The study found that the annealing treatment was beneficial to improve the crystallinity, the grain size, the transmittance of the films, and to reduce the resistivity of the films. When the ICO films were annealed at 210 °C in air for 50 min, the mobility of 93.3 cm<sup>2</sup>/V s, a carrier concentration of  $2.48 \times 10^{20}$ /cm<sup>3</sup>, an average transmittance of 86.5%, and the resistivity of  $2.6 \times 10^{-4}$  Ω cm were obtained. The ICO films presented higher carrier mobility and lower carrier concentration than the Sn-doped In<sub>2</sub>O<sub>3</sub> (ITO) films, which are conducive to improving the efficiency of solar cells.

## 1 Introduction

Recently, Ce-doped In<sub>2</sub>O<sub>3</sub> (ICO) transparent conductive oxide (TCO) films [1–3] have attracted more and more attention because of their higher carrier mobility ( $\mu_e$ ), lower resistivity ( $\rho$ ), and higher optical transmittance [4–8]. In 2014, Kang et al. [9] used magnetron sputtering to prepare Sn-doped In<sub>2</sub>O<sub>3</sub> (ITO) film and ICO film. They studied the influence of film thickness and Ce concentration on film properties. Since the Ce–O bond enthalpy

( $795 \pm 8$  kJ/mol) is higher than the In–O bond enthalpy ( $320.1 \pm 41.8$  kJ/mol) and the Sn–O bond enthalpy ( $531.8 \pm 12.6$  kJ/mol), the ICO film is easier to nucleate than the ITO film during the deposition process. Ce doping changes the growth mode of the film [9]. By optimizing the parameters, the best  $\mu_e$  of the ICO film is  $\sim 50$  cm<sup>2</sup>/V s. In 2016, An et al. [10] studied the influence of substrate temperature and found that the optical and electrical properties of ICO were better than ITO films. As the substrate temperature increases, the ICO film changes from

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amorphous to polycrystalline. When the substrate temperature is 150 °C, the highest mobility of ICO film is 84.3 cm<sup>2</sup>/V s. They applied ICO film instead of ITO film onto perovskite and silicon tandem cells, and the cell efficiency increased from 18.85% to 20.37%. In 2018, Shi et al. [11] used reactive plasma deposition (RPD) to prepare ICO films. They studied the influence of ICO film thicknesses on carrier concentration ( $N_e$ ) and  $\mu_e$ . It is found that as the film thickness increases, the  $N_e$  increases, and the  $\mu_e$  first increases and then decreases. When the film thickness is 30 nm, the maximum  $\mu_e$  is 153.7 cm<sup>2</sup>/V s, and the  $N_e$  is  $1.9 \times 10^{20}$ /cm<sup>3</sup>. They believe that when the film thickness is less than 30 nm, the  $\mu_e$  is affected by the surface scattering mechanism, and when the film thickness is thicker than 50 nm, the  $\mu_e$  is affected by the grain boundary scattering mechanism. In 2019, Mei et al. [12] used magnetron sputtering to prepare ICO films and studied the effects of Ce doping with different concentrations on the optical and electrical properties of TCO films. They found that Ce doping can change the nucleation and growth processes of TCO film. The grain growth of ICO film is oriented along the [111], and as the deposition time increases, the film tends to grow along the [100]. In addition, Ce doping can weaken crystallinity, reduce surface roughness, and increase the density of TCO film. Surface scattering is dominant in amorphous ICO, and ionized impurity scattering is dominant in polycrystalline ICO. ICO film shows excellent performance, which is worth studying.

In fact, most TCO films are prepared by physical vapor deposition (PVD) [13–15]. However, TCO films prepared by RPD with denser structures, higher crystallinity, smoother surface, higher conductivity, and better optical transmittance [16, 17]. In this paper, ICO films were prepared by RPD, and the effects of annealing temperature and annealing time on the electrical properties, optical properties, and film structure of ICO films were studied. By optimizing the experimental parameters, the film with the best performance was obtained, and the mechanism of annealing temperature and annealing time was explored.

## 2 Experimental

ICO films with a thickness of about 100 nm were prepared on glass substrates by RPD (Archers RPD-1Gun) with the targets composed of 97.0% In<sub>2</sub>O<sub>3</sub>: 3.0% CeO<sub>2</sub> in weight. The film deposition conditions were as follows: O<sub>2</sub> concentration (the percentage of O<sub>2</sub> in the total amount of O<sub>2</sub> and Ar) 18.5%, the current density of the plasma gun 160 A/m<sup>2</sup>, and the deposition pressure 1.5 mTorr. The samples were prepared at room temperature and then annealed in air. When studying the effect of annealing temperature, the fixed annealing time was 50 min, and the annealing temperature was 170–250 °C. When the effect of annealing time was studied, the fixed annealing temperature was 210 °C, and the annealing time was 10–90 min.

The optical transmittance of ICO films was measured in the wavelength range of 300–1200 nm via ultraviolet–visible spectrophotometer (UV-2600), the thickness was obtained via ellipsometer (Raditech SE-950), the  $\rho$ , the  $N_e$  and the  $\mu_e$  were acquired via Hall measurement (Swin HALL8800-II). Further, the film crystallinity was analyzed via X-ray diffractometer (XRD D8 ADVANCE).

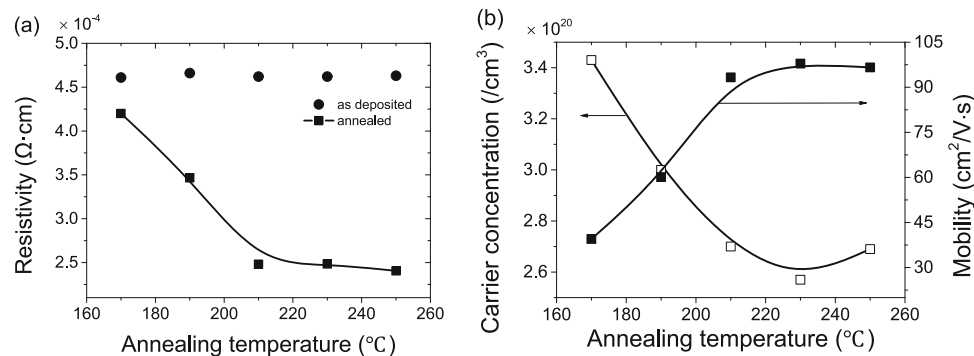
## 3 Results and discussion

### 3.1 The effect of annealing temperature on the properties of ICO films

Figure 1 shows the effect of annealing temperature on the electrical properties of ICO films. We found that the as-deposited films had a high  $\rho$  of  $\sim 4.6 \times 10^{-4} \Omega \text{ cm}$ , but the  $\rho$  decreased significantly after annealing with the  $\rho$  of  $\sim 2.5 \times 10^{-4} \Omega \text{ cm}$ . It is consistent with the results reported in the literature [18]. However, in our previous study [19], the lowest  $\rho$  of ITO is  $\sim 9.6 \times 10^{-4} \Omega \text{ cm}$ , which shows that ICO has better electrical performance. With the increase of annealing temperature, the  $\rho$  decreased more obviously. The change in  $\rho$  is determined by the  $N_e$  and  $\mu_e$ .  $\rho$  is calculated according to the following Eq. (1).

$$\rho = \frac{1}{q\mu_e N_e} \quad (1)$$

where,  $q$  is the quantity of electric charge ( $q = 1.6 \times 10^{-19} \text{ C}$ ). When the annealing temperature



**Fig. 1** **a** The resistivity ( $\rho$ ) of ICO films obtained before and after annealing as functions of the annealing temperature, the circle represents the  $\rho$  of the as-deposited film, and the square represents

the  $\rho$  of the film after annealing, **b** the carrier concentration ( $N_e$ ) and the carrier mobility ( $\mu_e$ ) of the annealed ICO films as functions of the annealing temperature

is lower, the  $\mu_e$  is lower, which results in higher  $\rho$ . As the annealing temperature increases, the increase in  $\mu_e$  is larger than the decrease in  $N_e$ , resulting in a decrease in  $\rho$ . The  $N_e$  is related to the concentration of oxygen vacancies [16, 20, 21] and activated  $\text{Ce}^{4+}$  [22]. Studies have shown that [23] there is a dynamic process of adsorption and desorption of oxygen atoms in the film when annealed in air. When the number of oxygen atoms adsorbed in the film is more than the number of oxygen atoms removed, the oxygen vacancies in the film decrease, resulting in a decrease in the  $N_e$ . Conversely, the  $N_e$  increases. As the annealing temperature increases, the number of oxygen atoms in the air entering the film increases, resulting in a decrease in the  $N_e$ . However, with the increase of annealing temperature, the number of activated  $\text{Ce}^{4+}$  increased, increasing in the  $N_e$ . In general, the  $N_e$  increases slightly at higher temperatures because of the increase of activated  $\text{Ce}^{4+}$ . At the same time, we found that as the annealing temperature increased, the  $\mu_e$  increased and then remained unchanged. When the annealing temperature is higher than  $210^{\circ}\text{C}$ , the  $\mu_e$  reaches about  $95 \text{ cm}^2/\text{V} \cdot \text{s}$ .

Figure 2 shows the effect of annealing temperature on the optical properties of ICO films. The average transmittance of the film was calculated from the transmittance spectrum in the wavelength range of 400–1100 nm. It can be seen that when the annealing temperature is higher than  $190^{\circ}\text{C}$ , the average transmittance of the film increases with the increase of the annealing temperature, and the average transmittance is higher than 85%. And from the transmittance spectrum, the transmittance in the full spectrum range increases with the increase of the annealing temperature. The transmittance is affected

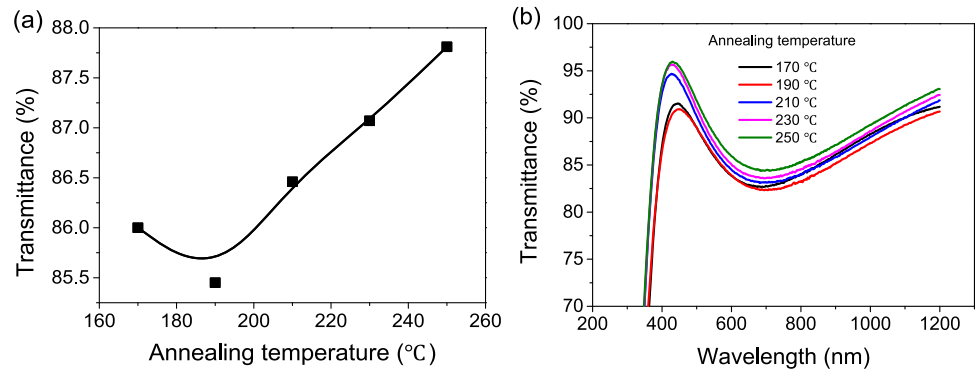
by the thickness and  $N_e$ . From the above results, the  $N_e$  decreases with the increase of annealing temperature, so less free-carrier absorption leads to an increase in the transmittance. However, a slight increase in thickness leads to a decrease in transmittance at  $190^{\circ}\text{C}$ .

In order to explore the effect of annealing temperature on the structure of ICO films, we analyzed the film crystallinity via XRD. Figure 3a shows that the diffraction peaks of each sample at different annealing temperatures are consistent with the standard XRD spectrum of  $\text{In}_2\text{O}_3$  (JCPDS No. 06-0416). It shows that no new compound is formed, and the film is polycrystalline. As the annealing temperature increases, the diffraction peak intensity of the (222) plane increases, indicating that the crystallinity of the film is improved. It can be seen from Fig. 3b that as the annealing temperature increases, the center position of the diffraction peak of the (222) crystal plane moves to a large angle, and when the annealing temperature is  $250^{\circ}\text{C}$ , it moves to a small angle. The change of the center position of the diffraction peak mainly reflects the change of the stress in the film [24, 25]. Properly increasing the annealing temperature is beneficial to reduce the internal stress of the film, but when the annealing temperature is too high, the film stress increases. Figure 3c shows the grain size ( $D$ ) and the full width at half maximum (FWHM). The  $D$  corresponding to the (222) diffraction peak was quantitatively characterized according to the Scherrer formula [19].

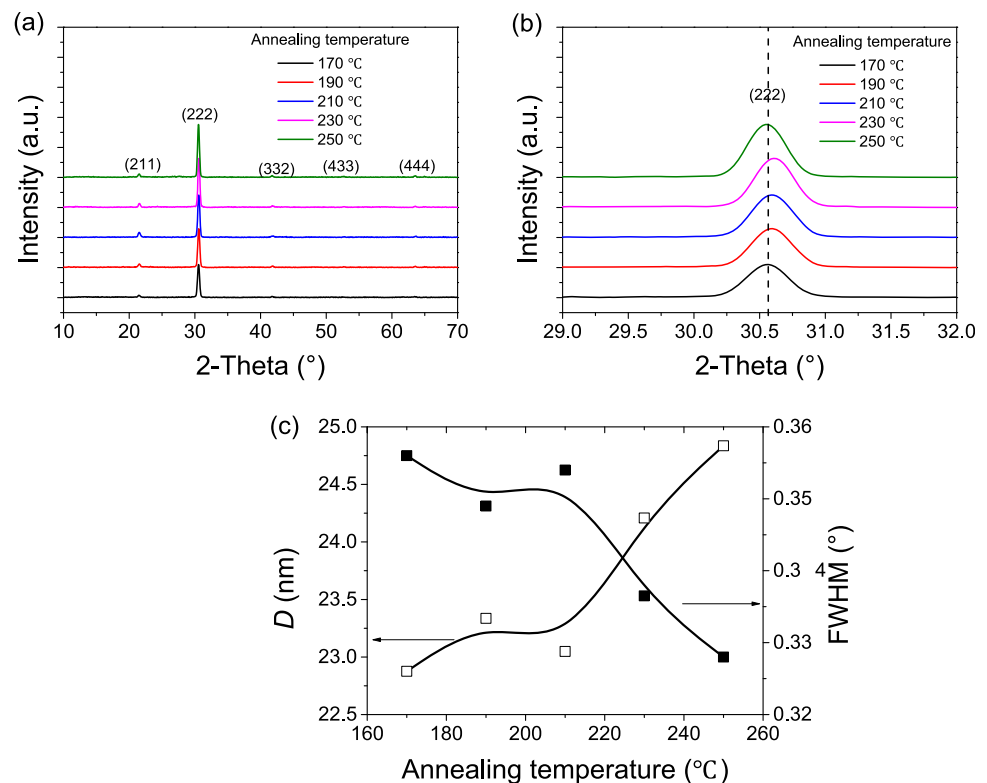
$$D = \frac{K\gamma}{B \cos \theta} \quad (2)$$

where,  $K$  is the Scherrer constant ( $K = 0.89$ ),  $B$  is the FWHM of the diffraction peak,  $\theta$  is the diffraction

**Fig. 2** **a** The average transmittance of the annealed ICO films as a function of different annealing temperatures, **b** the transmittance spectrum of the annealed ICO films as a function of wavelength for different annealing temperatures



**Fig. 3** **a** XRD curves of the annealed ICO films obtained with different annealing temperatures, **b** XRD curves of (222) crystal plane, **c** the corresponding grain size ( $D$ ) and the full width at half maximum (FWHM) calculated for the ICO films



angle, and  $\gamma$  is the X-ray wavelength ( $\gamma = 1.54 \text{ \AA}$ ). As the annealing temperature increases, the  $D$  increases and the FWHM decreases, which indicates that the crystallinity of the film is improved. The  $\mu_e$  mainly depends on lattice vibration, neutral impurity scattering, ionized impurity scattering, and grain boundary scattering [26, 27]. The increase of  $D$  is beneficial to reduce the grain boundary scattering and improve the  $\mu_e$ . This result is also consistent with the Hall result above.

Considering the influence of annealing temperature on the electrical properties, optical properties, film structure, and fabrication cost of ICO films. The optimal annealing temperature is 210 °C, and the

obtained ICO film was with the  $\mu_e$  of  $93.3 \text{ cm}^2/\text{V s}$ , the  $N_e$  of  $2.48 \times 10^{20}/\text{cm}^3$ , the average transmittance of 86.5%, and the  $\rho$  of  $2.6 \times 10^{-4} \Omega \text{ cm}$ , which is better than the ITO film with the  $\mu_e$  of  $24.1 \text{ cm}^2/\text{V s}$ , the  $N_e$  of  $2.7 \times 10^{20}/\text{cm}^3$ , the average transmittance of 86.0%, and the  $\rho$  of  $9.6 \times 10^{-4} \Omega \text{ cm}$  [19].

### 3.2 The effect of annealing time on the properties of ICO films

Figure 4 shows the effect of annealing time on the electrical properties of ICO films. As the annealing time increases, the  $\rho$  decreases continuously. When the annealing time is shorter, lower  $N_e$  and  $\mu_e$  lead to

higher  $\rho$ . As the annealing time increases, the number of desorbed oxygen atoms in the film is larger than the number of adsorbed oxygen atoms, which increases the oxygen vacancy concentration and the  $N_e$ . The energy accumulation in the film causes the concentration of activated  $Ce^{4+}$  to rise as annealing time rises, which leads to an increase in  $N_e$ . The  $\mu_e$  increases and then decreases with increasing annealing time. The main reason is that the increase of the activated  $Ce^{4+}$  leads to the enhanced scattering of ionized impurities and the decrease of the  $\mu_e$ . In addition to this, annealing for too long may cause lattice vibrations or introduce new defects [28], which reduce  $\mu_e$ .

Figure 5 shows the relationship between annealing time and ICO films' transmittance. When the annealing time is longer than 30 min, the average transmittance of the film increases with the increase of the annealing time. Meanwhile, from the transmittance spectrum, the transmittance of the long wavelength increases with the increase of annealing time. The transmittance of the thin film is related to the absorption and reflection of the film. With the change of film crystallinity, grain size, etc., the transmittance of the film changes.

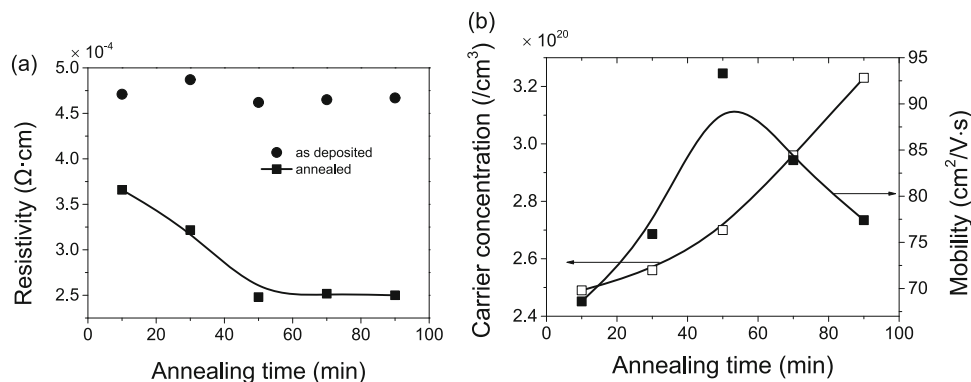
We also analyzed the film crystallinity via XRD, as shown in Fig. 6a, there is no new material generated after annealing and the film is polycrystalline. The diffraction peak intensity of the (222) plane increases with the annealing time increases, indicating that the crystallinity of the film is improved. It can be seen from Fig. 6b that as the annealing time increases, the center position of the diffraction peak of the (222) crystal plane moves to a large angle, which indicates

that the internal stress of the film is released. Figure 6c shows the relationship between the  $D$  or the FWHM and the annealing time. As the annealing time increases, the  $D$  increases, resulting in decreased grain boundary scattering and increased  $\mu_e$ . With the increase of annealing time, the FWHM decreases, which also indicates that the crystallinity of the film is improved.

From the perspective of high  $\mu_e$  and low  $N_e$ , ICO films with better performance can be obtained with an annealing time of 50 min. Under these conditions, the  $\mu_e$  of the ICO film was  $93.3 \text{ cm}^2/\text{V s}$ , the  $N_e$  was  $2.48 \times 10^{20}/\text{cm}^3$ , the average transmittance was 86.5%, and the  $\rho$  was  $2.6 \times 10^{-4} \Omega \text{ cm}$ .

## 4 Conclusion

It was found that the annealing temperature and annealing time significantly affected the electrical properties, optical properties, and film structure of ICO films. During the annealing process, there is recombination, recrystallization, and film stress relief. As the annealing temperature increases, the crystallinity of the film improves and the  $D$  increases, resulting in decreased grain boundary scattering, increased  $\mu_e$ , and decreased  $\rho$ . At the same time, as the temperature increase, the  $N_e$  decreases, which improves the transmittance of the film. However, when the annealing temperature is too high, other defects may be caused. As the annealing time increases, the film stress is released, the crystallinity improves, and the  $D$  increases, resulting in decreased grain boundary scattering and increased  $\mu_e$ . However, when the annealing time is too long, more

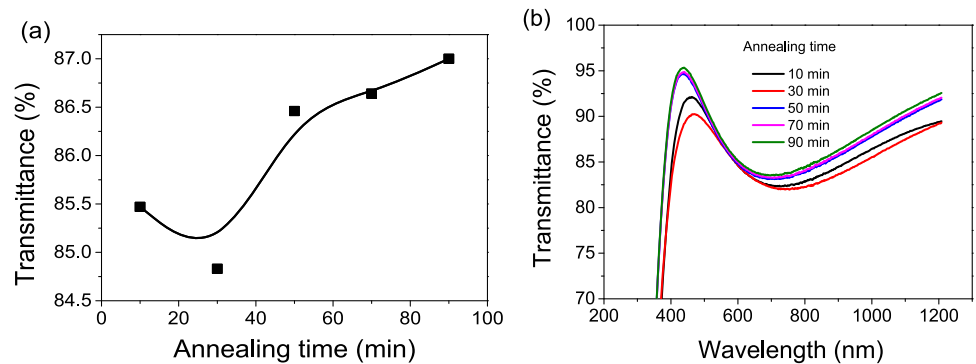


**Fig. 4** **a** The resistivity ( $\rho$ ) of ICO films obtained before and after annealing as functions of the annealing time, the circle represents the  $\rho$  of the as-deposited film, and the square represents the  $\rho$  of

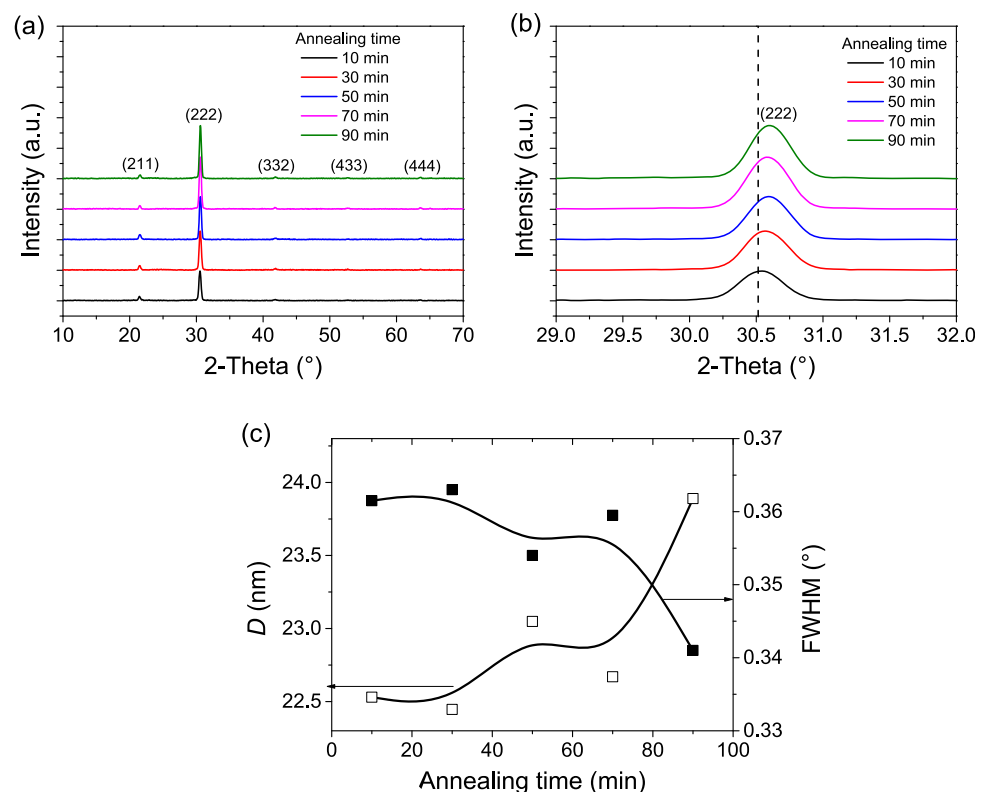
the film after annealing, **b** the carrier concentration ( $N_e$ ) and the carrier mobility ( $\mu_e$ ) of the annealed ICO films as functions of the annealing time



**Fig. 5** **a** The average transmittance of the annealed ICO films as a function of different annealing times, **b** the transmittance spectrum of the annealed ICO films as a function of wavelength for different annealing times



**Fig. 6** **a** XRD curves of the annealed ICO films obtained with different annealing times, **b** XRD curves of (222) crystal plane, **c** the corresponding grain size ( $D$ ) and full width at half maxima (FWHM) calculated for the ICO films



oxygen atoms are desorbed or more activated  $\text{Ce}^{4+}$  in the film, resulting in a higher  $N_e$ . There are potential barriers at the grain boundaries that trap free carriers and reduce  $\mu_e$ . By optimizing the annealing process, we obtain the best performance ICO films after annealing at 210  $^{\circ}\text{C}$  in air for 50 min. The  $\mu_e$  of the ICO film was  $93.3 \text{ cm}^2/\text{V s}$ , the  $N_e$  was  $2.48 \times 10^{20}/\text{cm}^3$ , the average transmittance was 86.5%, and the  $\rho$  was  $2.6 \times 10^{-4} \Omega \text{ cm}$ . ICO films have better performance than traditional ITO films for solar cells.

## Author contributions

Conceptualization: LZ; methodology: YG, LZ, WW; formal analysis and investigation: YG, HL; writing—original draft preparation: HL; writing—review and editing: HL, XJ, HD, LZ, WW; funding acquisition: LZ, WW, JZ; resources: WW, JZ; supervision: WW, JZ.

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## Data availability

All data generated or analysed during this study are included in this published article.

## Declarations

**Conflict of interest** The authors have no relevant financial or non-financial interests to disclose.

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