



Oregon State  
University

COLLEGE OF ENGINEERING

School of Mechanical, Industrial,  
and Manufacturing Engineering

# Laser directed energy deposition of Ni-based superalloy GRCop42 joints for multi-metal components

## PhD Final Examination

***Mechanical Engineering with Advanced Manufacturing Option***

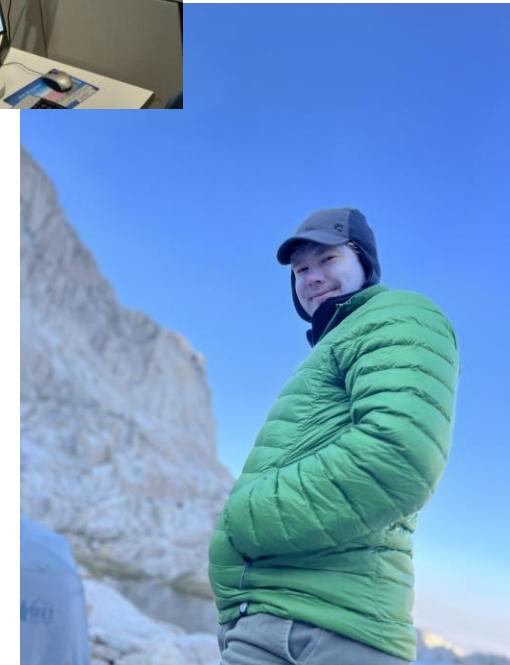
PhD Student: Jakub (Kuba) Preis

Advisor: Dr. Somayeh Pasebani

Committee: Dr. Brian Paul, Dr. Donghua Xu, Dr. Julie Tucker, Dr. David H McIntyre

# About the presenter

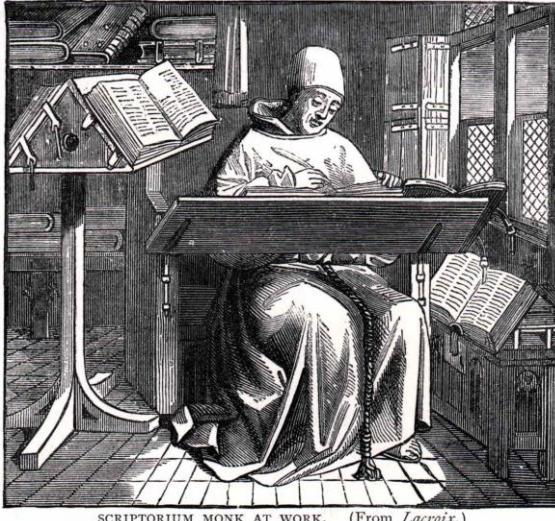
- 2015 - 2019      B.S. Manufacturing Engineering  
*California Polytechnic State University, San Luis Obispo, CA, USA*
- 2019 - 2020      Additive Manufacturing Engineer  
*Divergent Technologies, Los Angeles, CA, USA*
- 2020 - 2021      Additive Manufacturing Engineer  
*Divergent Technologies, on assignment to SLM solutions, Lubeck, Germany*
- 
- 2021 - Present    Graduate Research Assistant  
*Oregon State University, Corvallis, OR, USA*



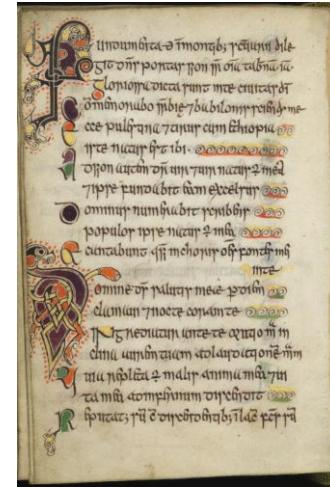
Hobbies: Biking, hiking, anything that has to do with the outdoors.

# Pre-Industrial printing/manufacturing

## 2 dimensional



SCRIPTORIUM MONK AT WORK. (From Lacroix.)



Scotland's Book of Kells, 800 AD

## 3 dimensional



### Disadvantages :

- Not repeatable (every book is different).
- Expensive and slow.
  - Every book must be copied/written by a trained expert.

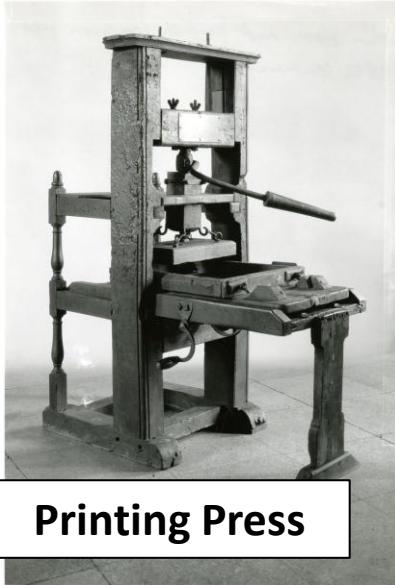
### Disadvantages :

- Not repeatable (every part is different).
- Expensive and slow.
  - Every part must be made by a trained expert.

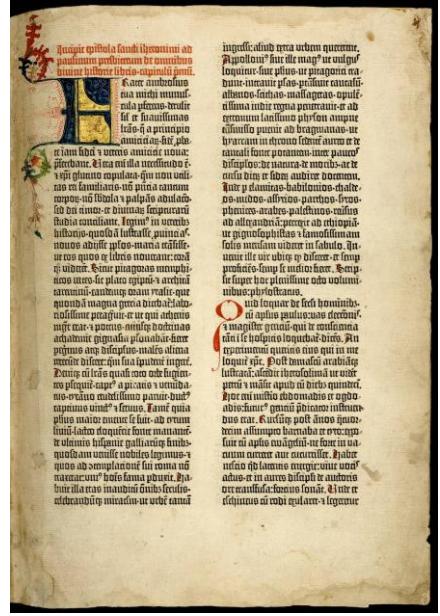
# Industrial printing/manufacturing



## 2 dimensional



Printing Press



Gutenberg bible

1455 AD

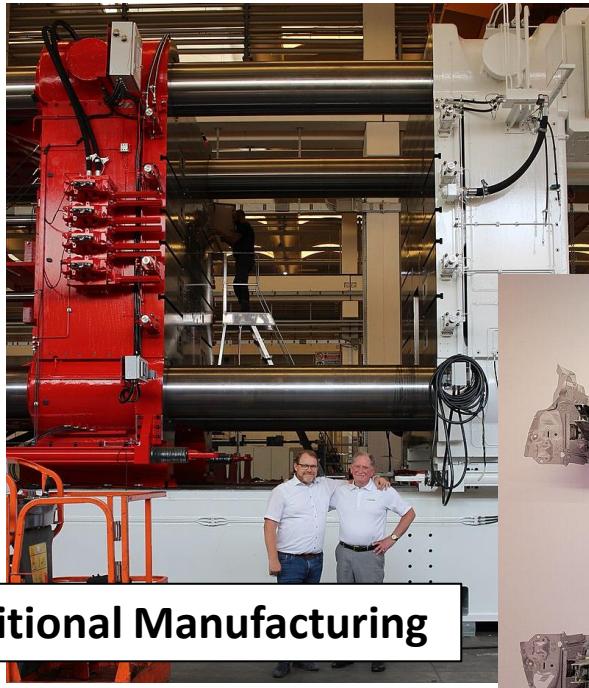
### Advantages:

- Repeatable.
- Faster than manual copying.

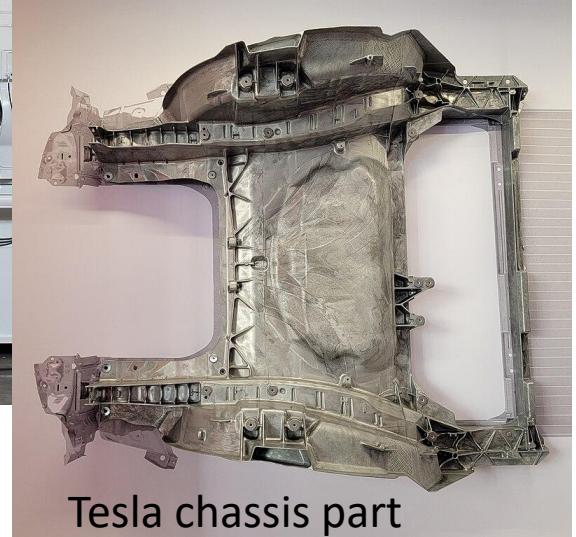
### Disadvantages:

- Each setup is page-specific.
- Cannot print all images.

## 3 dimensional



Traditional Manufacturing



Tesla chassis part

### Advantages:

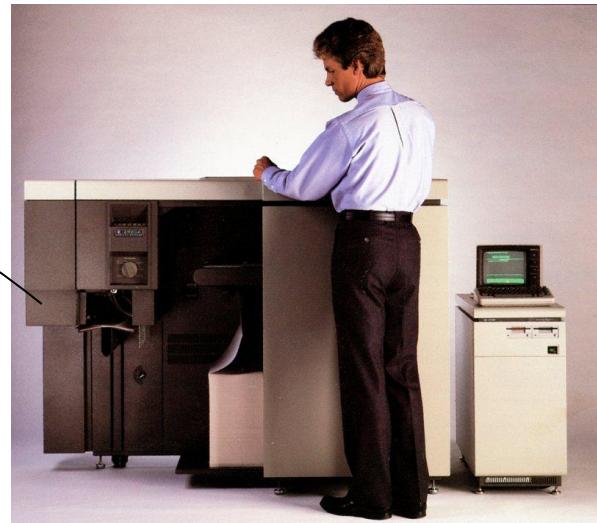
- Repeatable.
- Better quality.
- Faster than manual manufacturing.

### Disadvantages :

- Requires geometry specific tooling (large fixed cost).
- Cannot fabricate all geometries (assemblies required).

# Modern printing/manufacturing

## 2 dimensional



Inkjet printer (black and white)

### Advantages:

- Do not need to change configuration each page.

### Disadvantages :

- Slow.
- Only black and white.

## 3 dimensional



Additive  
Manufacturing (AM)



GE fuel nozzle

### Advantages:

- Greater design freedom (can print more geometries).
- No tooling, machine is not geometry specific.

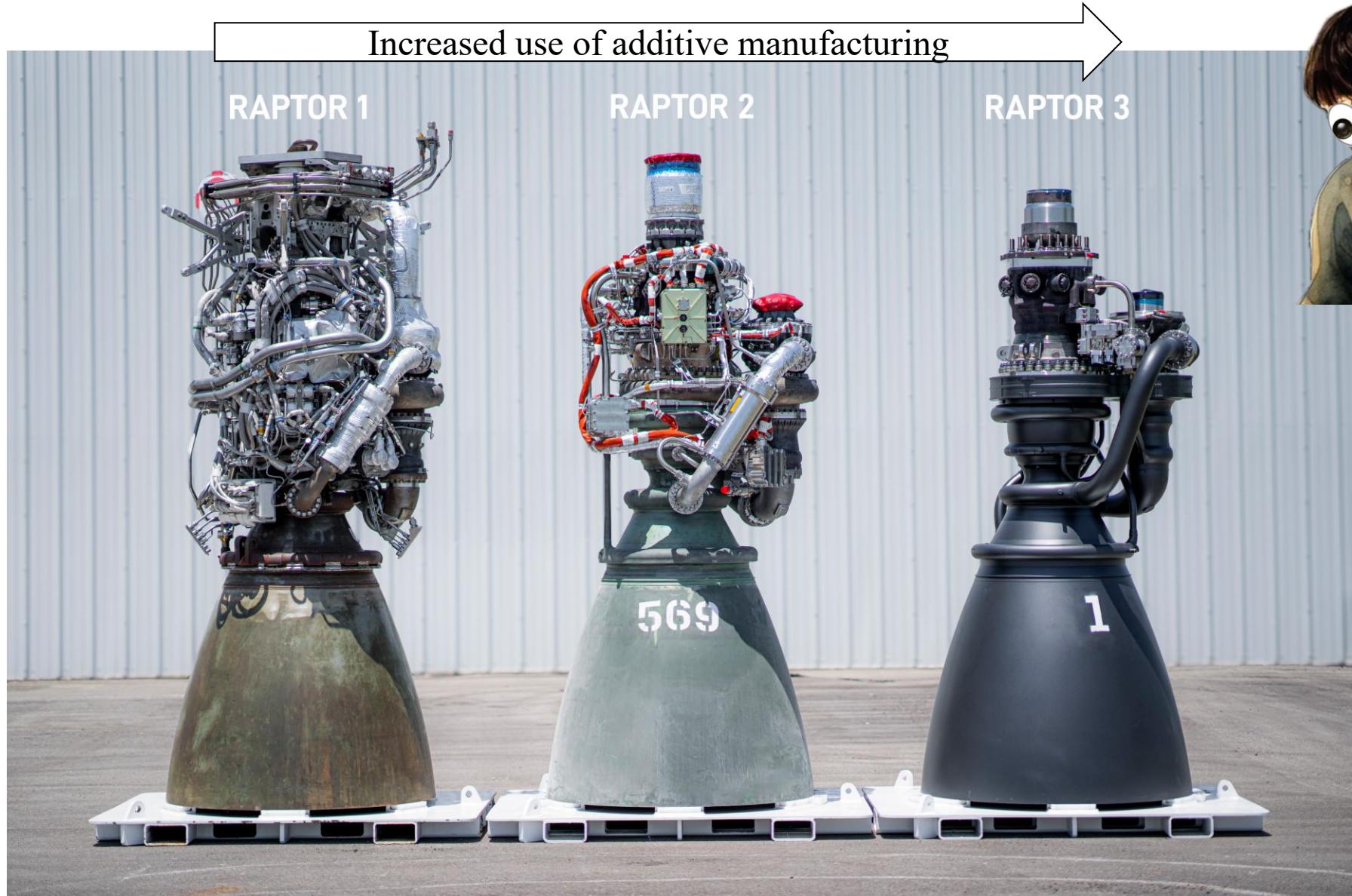
### Disadvantages :

- Slow.
- Constrained to one material.

# Advantages of AM over traditional manufacturing



Increased use of additive manufacturing

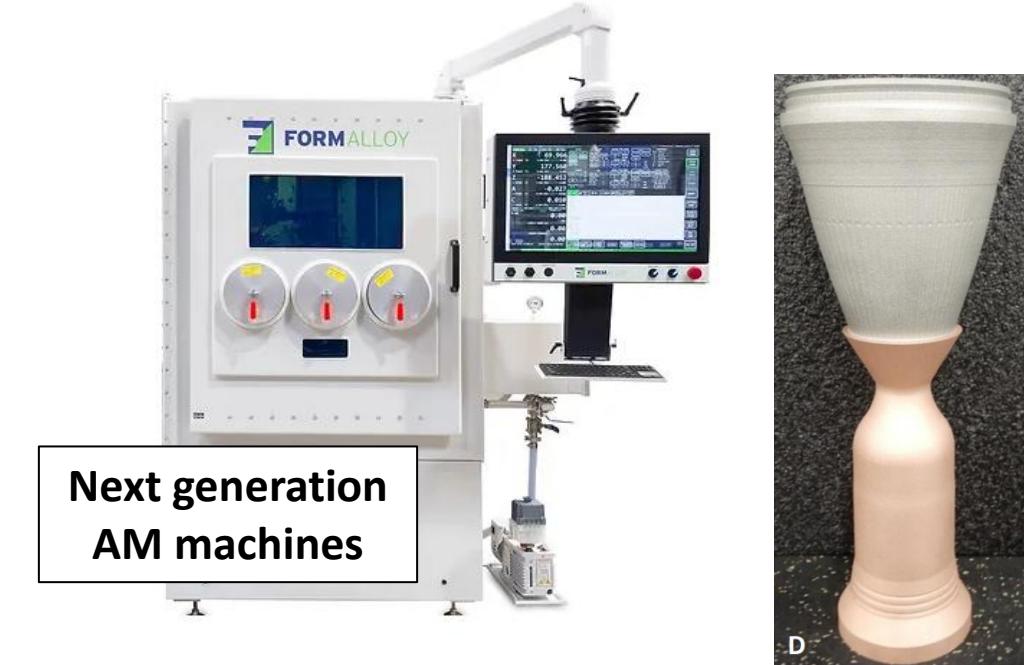


# Next generation printing/manufacturing

2 dimensional



3 dimensional



Rocket Exhaust Chamber + Nozzle (NASA)

## Advantages:

- Can essentially print any design.
- **Can print color.**

## Advantages:

- Higher deposition rate (faster printing).
- Multi-material capability.

## Constraints:

- Part resolution.
- **Metallurgical constraints.**

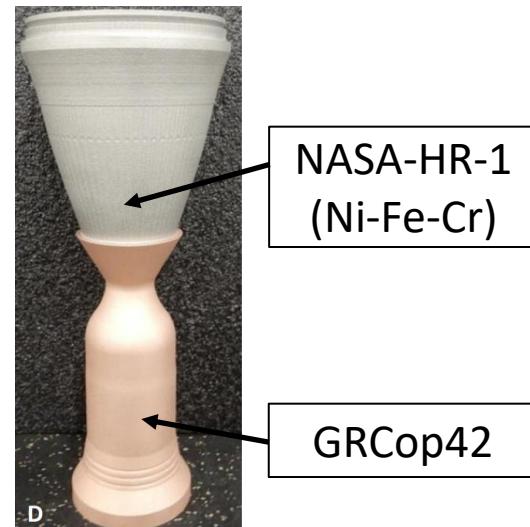
# Thermal management of Ni-based superalloys

- The joining of Inconel superalloys to high temp Cu alloys is of interest in applications which require both regions of **high temperature strength** and **high thermal conductivity**.

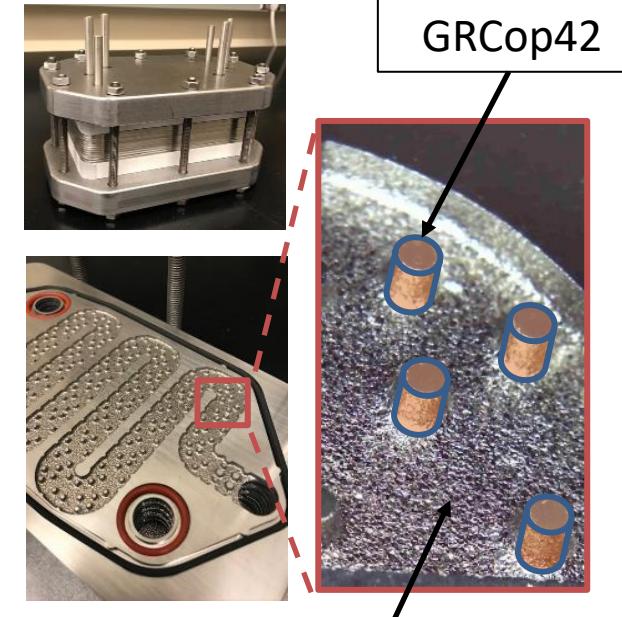
**Rocket Exhaust Chamber [1]**  
*NASA Marshall Space Flight Center*



**Rocket Exhaust Chamber + Nozzle [2]**  
*NASA Marshall Space Flight Center*

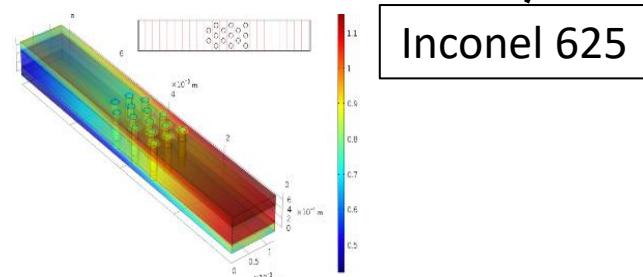


**Superheater Concept**  
*Oregon State University*



Alloy	Thermal Conductivity [W/mK]	UTS, 25 C [MPa]	UTS, 650 C [MPa]
Inconel 625	10 [5]	1121 [4]	637 [6]
GRCop42	275 [3]	360 [3]	85 [1]

- [1] Paul R. Gradi, Christopher S. Protz, David L. Ellis, and Sandy E. Greene. "Progress in additively manufactured copper-alloy GRCOP-84, GRCOP-42, and bimetallic combustion chambers for liquid rocket engines". In: Proceedings of the International Astronautical Congress, IAC. Vol. 2019-October. 2019.  
[2] Paul R. Gradi, Chris Protz, John Fikes, Sandi Miller, Laura Evans, Sandi Clark, Laura Evans, Sandi Miller, David Ellis, and Tyler Hudson. "Lightweight thrust chamber assemblies using multi-alloy additive manufacturing and composite over-wrap". In: AIAA Propulsion and Energy 2020 Forum. 2020.  
[3] Paul R. Gradi, Christopher S. Protz, David L. Ellis, and Sandy E. Greene. "Progress in additively manufactured copper-alloy GRCOP-84, GRCOP-42, and bimetallic combustion chambers for liquid rocket engines". In: Proceedings of the International Astronautical Congress, IAC. Vol. 2019-October. 2019.  
[4] MD Ashabul Anam. "Microstructure and mechanical properties of selective laser melted superalloy Inconel 625." In: Elsevier (2018).  
[5] Asit Kumar Parida and Kalipada Maiti. "Comparison the machinability of Inconel 718, Inconel 625 and Monel 400 in hot turning operation". In: Engineering Science and Technology, an International Journal 21.3 (June2018), pp. 364–370. issn: 2215-0986. doi:10.1016/J.JESTCH.2018.03.018.11  
[6] : Kim K-Sik, Kang T-Hoon, Kassner ME, Son K-Tae, Lee K-Ahn, High-Temperature Tensile and High Cycle Fatigue Properties of Inconel 625 Alloy Manufactured by Laser Powder Bed Fusion, Additive Manufacturing (2020), doi: <https://doi.org/10.1016/j.addma.2020.101377>



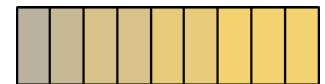
# Outline

**Goal:** Understand the process-structure-property relationship in Inconel – GRCop42 joints

- Impact of parameters for LDED processing of wire Inconel 625
  - ❖ Manuscript 1: *Preis et al., Influence of travel speed on microstructure and mechanical behavior of Inconel 625 fabricated using wire fed laser directed energy deposition, Journal of Materials Processing Technology*
- Impact of Inconel 625 GRCop42 mixing
  - ❖ Manuscript 2: *Preis et al., Effect of Liquid Miscibility Gap on Defects in Inconel 625–GRCop42 Joints through Analysis of Gradient Composition Microstructure, Journal of Manufacturing and Materials Processing*
- Bimetallic joining of Inconel 625 with GRCop42 using LDED
  - ❖ Manuscript 3: *Preis et al., Effect of laser power and deposition sequence on microstructure of GRCop42 - Inconel 625 joints fabricated using laser directed energy deposition, Materials and Design*
- Development of a filler composition for joining of Inconel 718 with GRCop42 and LDED printing
  - ❖ Manuscript 4: *Microstructure, microhardness, and thermal properties of Inconel 718 - GRCop42 joint with liquid miscibility gap avoiding filler compositions*

Inconel 625

Inconel 625 GRCop42 mixtures



GRCop42  
Inconel 625

GRCop42  
Filler  
Inconel 718

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Inconel 625

Inconel 625 GRCop42

mixtures



GRCop42

Inconel 625

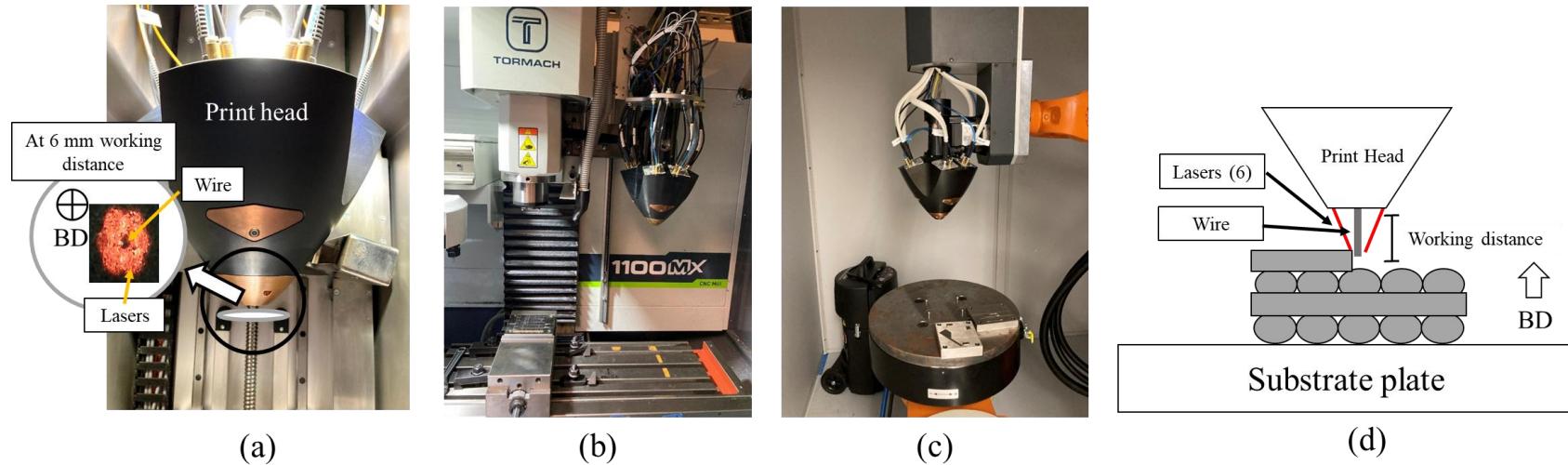
GRCop42

Filler

Inconel 718

# Key terms

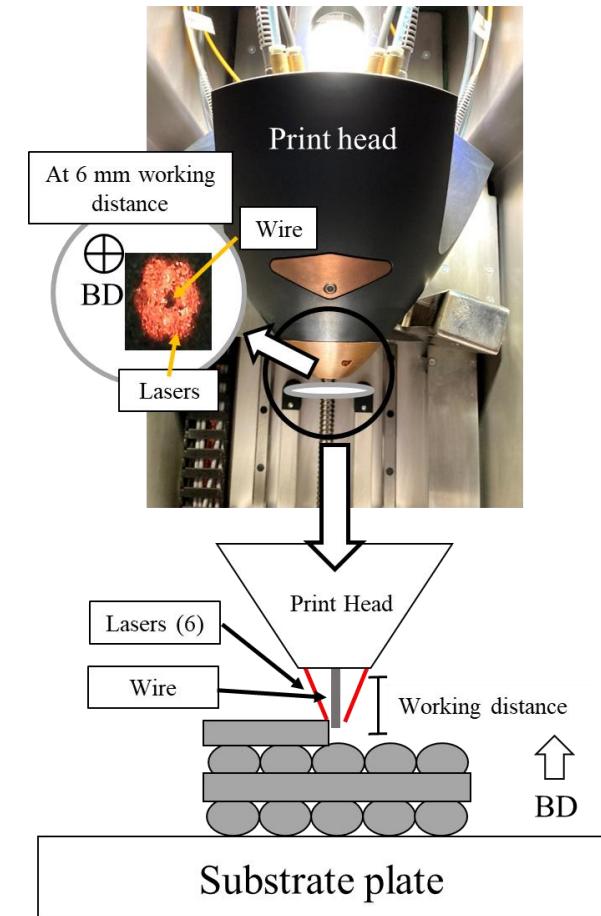
- Directed Energy Deposition



- Travel speed – the linear speed (typically in mm/min) at which a laser exposes a given pass.
- Laser power – the power (typically in Watts) used during the exposure of a given pass.

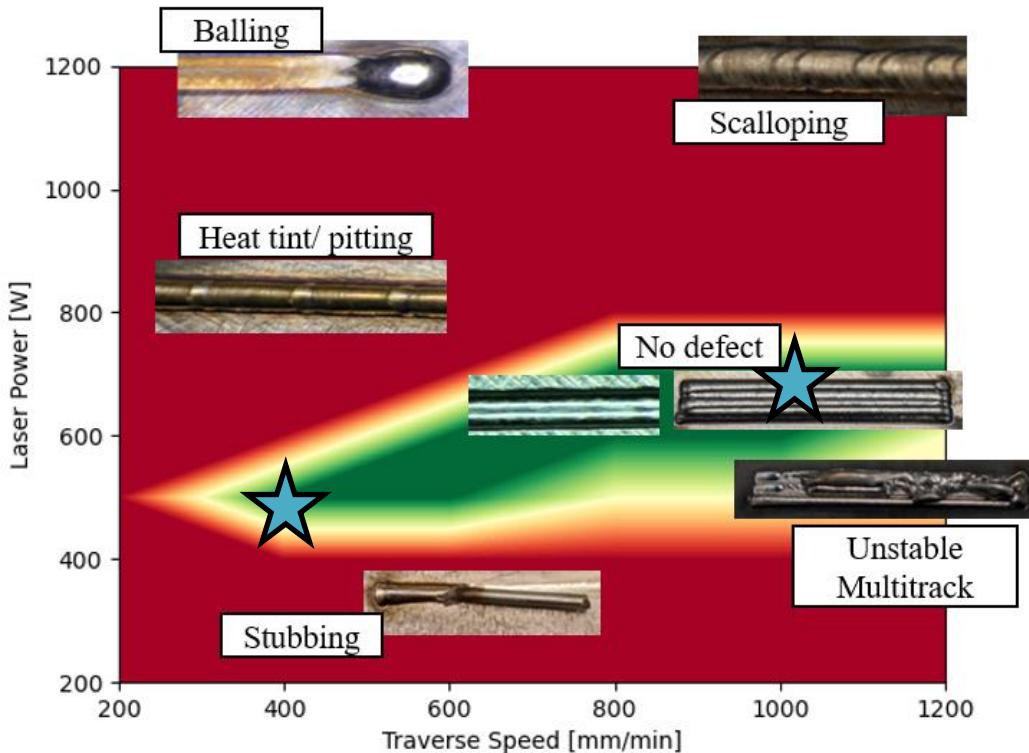
# Wire LDED of Inconel 625

- Meltio M450 wire-power system used
  - Advantages of wire LDED: 100% of material utilized, relatively fast print speed
- Processing map developed for single track deposition.
- A high-travel-speed and low-travel speed parameter selected for microstructure and mechanical property characterization.



Alloy	Ni	Cr	Nb	Ta	Mo	Cu	Fe	Ti	Al	Other
Inconel 625	64.8	22.2		3.49	8.6	.016	0.26	0.21	0.14	.284

# Single and Multi Track Experiments

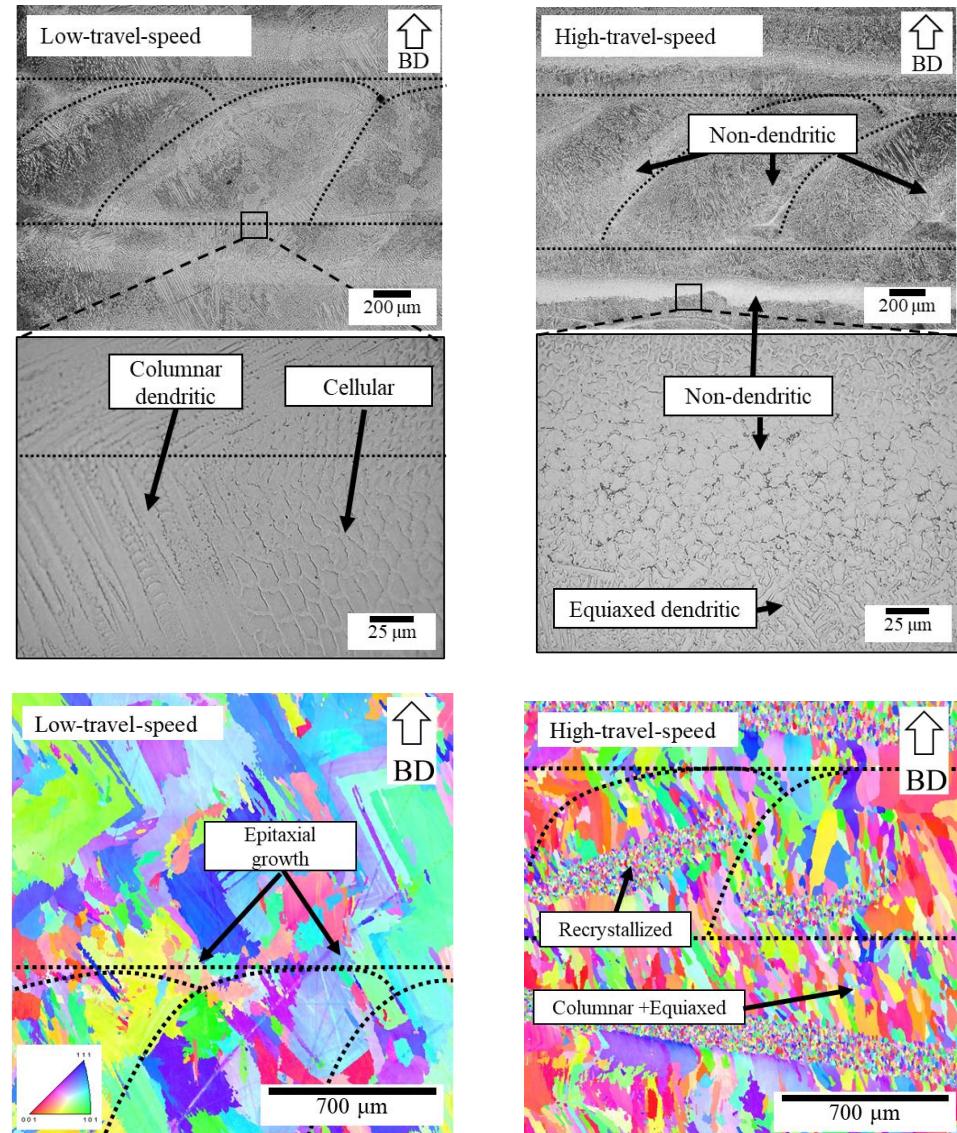


- Parameter space of defect free single and multi tracks shown in **green**.
- Multi track defect mode consisted of **inability to change direction** and occurred at **high speed**.
- A overlap ration (ratio of hatch spacing to bead width) of 0.7 found to be optimal for multi tracks.
- **High-travel-speed** and **low-travel speed** parameters identified for further characterization.

Parameter	Laser Power W	Travel Speed mm/min
Low-travel-speed	500	400
High-travel-speed	700	1000

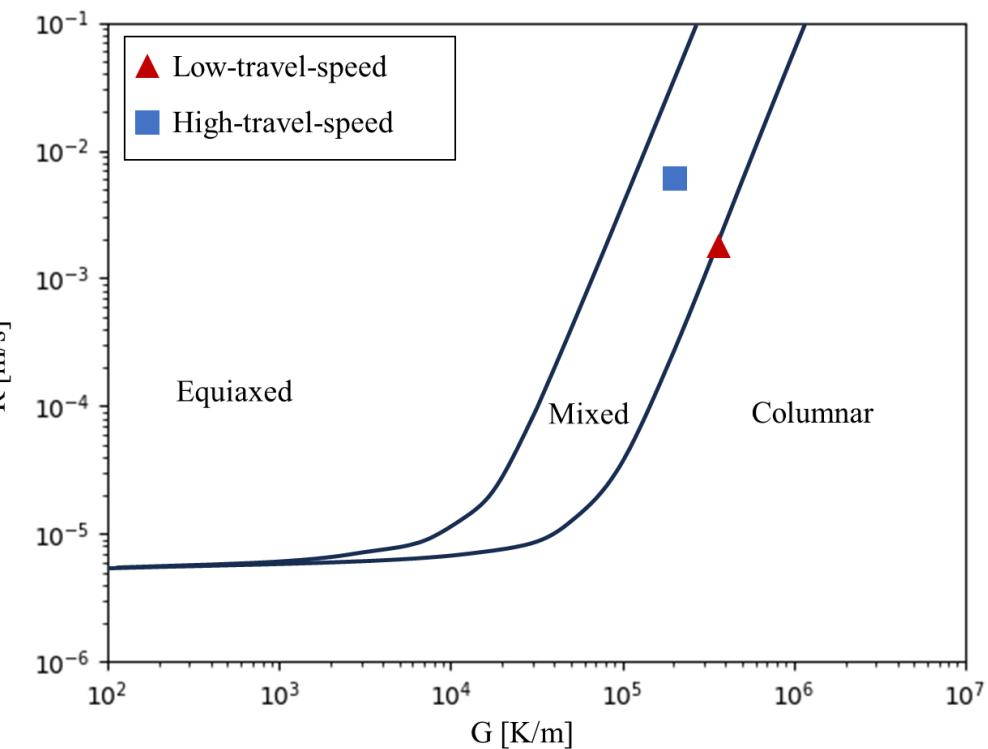
# Microstructural characterization

- **Columnar grains** observed in the low-travel-speed sample.
- **Columnar + equiaxed grains** observed in the high-travel-speed sample.
- Epitaxial growth observed in both samples.



# Influence of parameters on microstructure

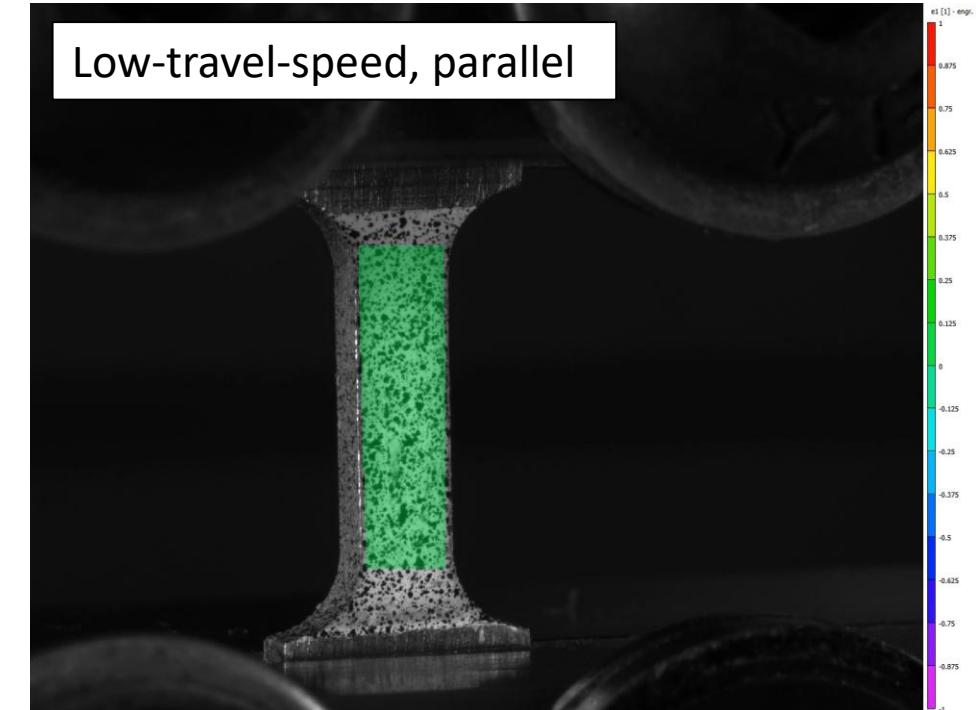
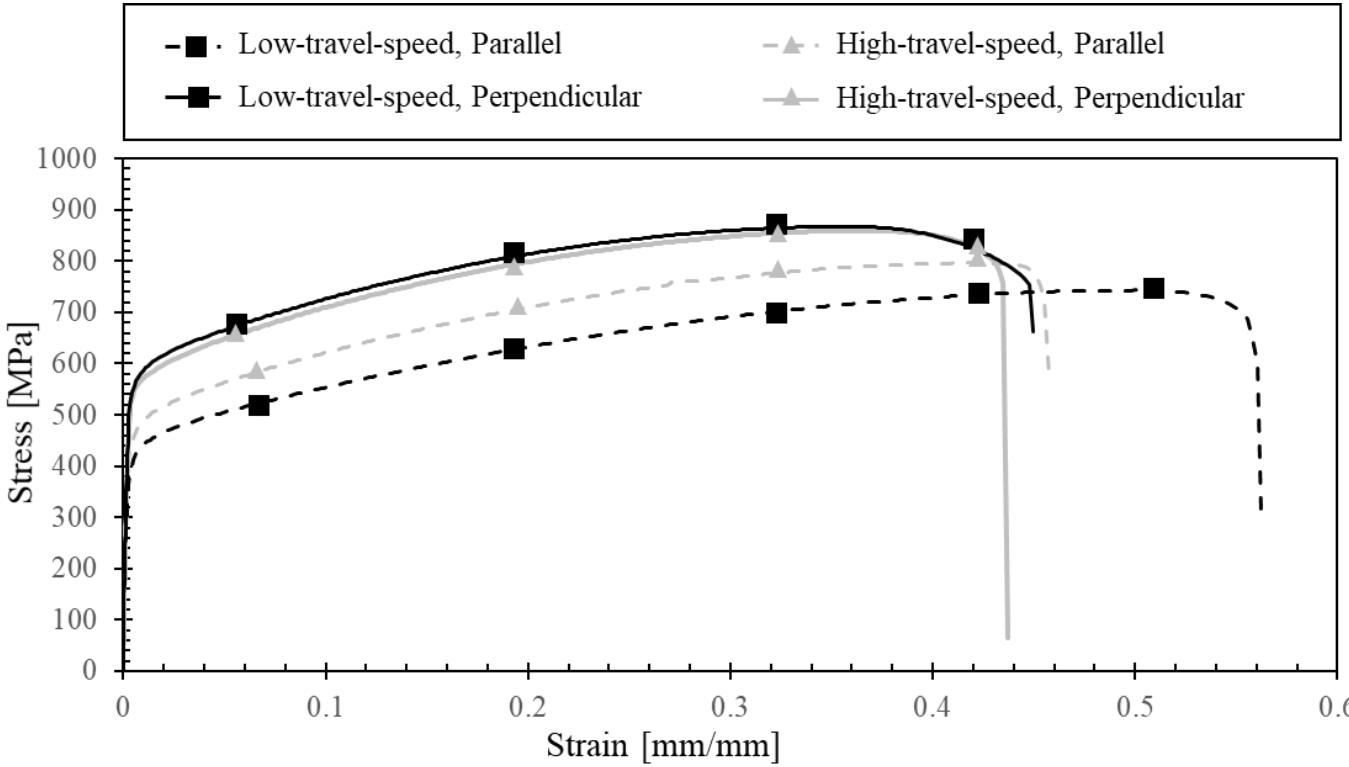
- Solidification rate and thermal gradient calculated based on
  - Melt pool depth from thermal-fluid modeling
  - Melt pool length from thermal-fluid modeling
  - Cooling rate from experimental thermocouple data
- The high-travel-speed parameter has a lower thermal gradient and higher solidification rate.
- When plotted onto a microstructure selection chart, the calculated results are in agreement with experimental results.



Parameter	Low-travel-speed	High-travel-speed
Melt pool depth [mm]	0.87	0.8
Melt pool length [mm]	2.81	2.17
Cooling rate [K/s]	723	1249
Solidification rate [m/s]	$2.1 \times 10^{-3}$	$6.2 \times 10^{-3}$
Thermal gradient [K/m]	$3.5 \times 10^5$	$2.0 \times 10^5$

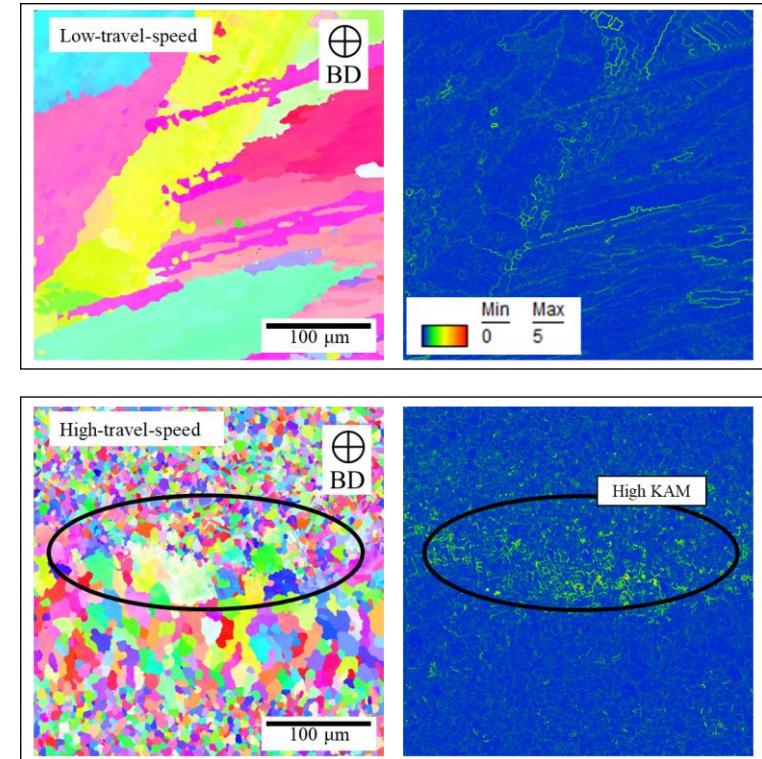
# Mechanical properties

- The high-travel-speed parameter shows a higher yield strength when pulled parallel to the build direction.
- Equivalent yield strength observed for both parameters when pulled perpendicular to the build direction.
- Higher yield strength observed for samples pulled perpendicular to the build direction.
- No difference in strain distribution observed between high travel speed and low travel speed parameters.



# Influence of microstructure on mechanical properties

- Hall-Petch strengthening calculated for each parameter and orientation based on EBSD.
- Dislocation strengthening calculated for each parameter and orientation based on KAM (from EBSD).
- Results suggest dislocation strengthening primarily contributes to observed differences in yield strength.
- Calculations inaccurate for high-travel-speed due to non-isotropic structure
  - Cannot assume in-plane homogeneity.



Orientation	Parameter	Hall-Petch strengthening MPa	Dislocation strengthening MPa	Total calculated MPa	Experimental MPa
Parallel	Low-travel-speed	0	0	0	0
Perpendicular	Low-travel-speed	30	105	135	134
Parallel	High-travel-speed	27	52	79	45
Perpendicular	High-travel-speed	28	70	98	135

# Outline

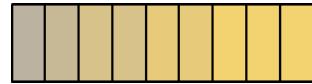
**Goal:** Understand the process-structure-property relationship in Inconel – GRCop42 joints

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Inconel 625

- Impact of Inconel 625 GRCop42 mixing
  - ❖ Manuscript 2: *Preis et al., Effect of Liquid Miscibility Gap on Defects in Inconel 625–GRCop42 Joints through Analysis of Gradient Composition Microstructure, Journal of Manufacturing and Materials Processing*

Inconel 625 GRCop42 mixtures



- Bimetallic joining of Inconel 625 with GRCop42 using LDED
  - ❖ Manuscript 3: *Preis et al., Effect of laser power and deposition sequence on microstructure of GRCop42 - Inconel 625 joints fabricated using laser directed energy deposition, Materials and Design*
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GRCop42  
Inconel 625

GRCop42  
Filler  
Inconel 718

## Key terms

- Liquid miscibility gap – two liquids cannot mix (lack of liquid state mixing).



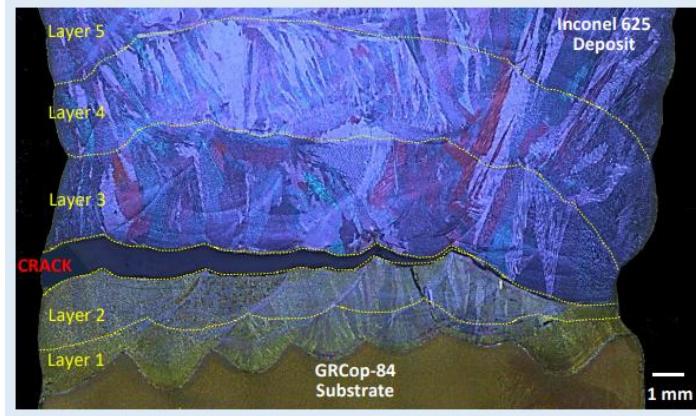
Example: olive oil and  
balsamic vinaigrette

- Brittle phase – a composition and corresponding crystal structure which cannot withstand elongation.

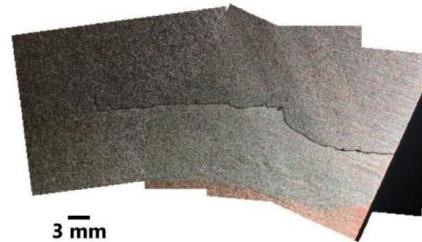
# Past work on Ni-superalloy GRCop joints

- Both **cracked** and **defect free** joints reported in the literature.
- No consensus as to the root cause and conditions under which cracking occurs – literature gap.

## Cracked Joints

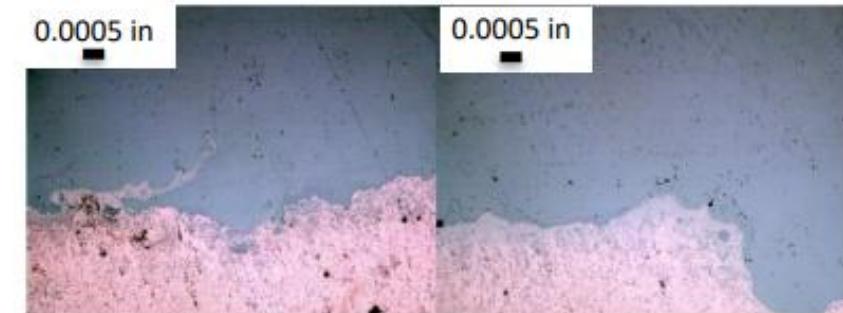


Wire Ebeam Inconel 625 onto GRCop84 [4]

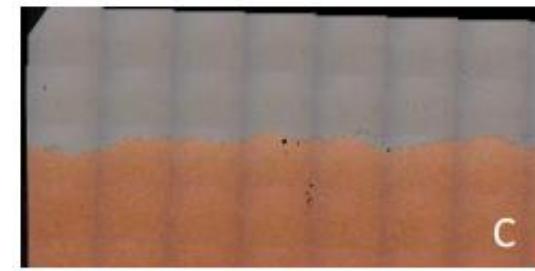


Wire Ebeam Inconel 625 onto GRCop42 [5]

## Defect Free Joints



Powder DED onto GRCop42 [2]



Wire EBeam onto GRCop84 [1]

[1] Progress in Additively Manufactured Copper-Alloy GRCop-84, GRCop-42, and Bimetallic Combustion Chambers for Liquid Rocket Engines

[2] Lightweight Thrust Chamber Assemblies using Multi-Alloy Additive Manufacturing and Composite Overwrap

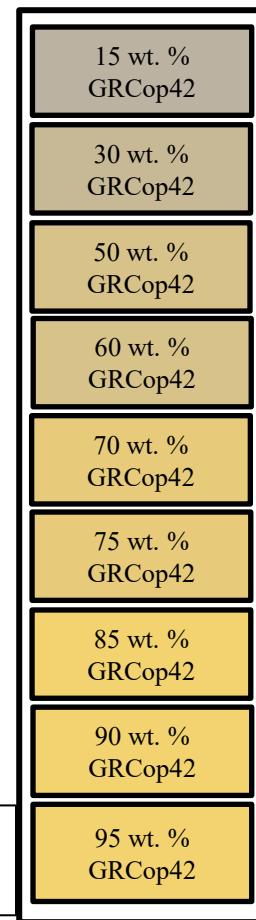
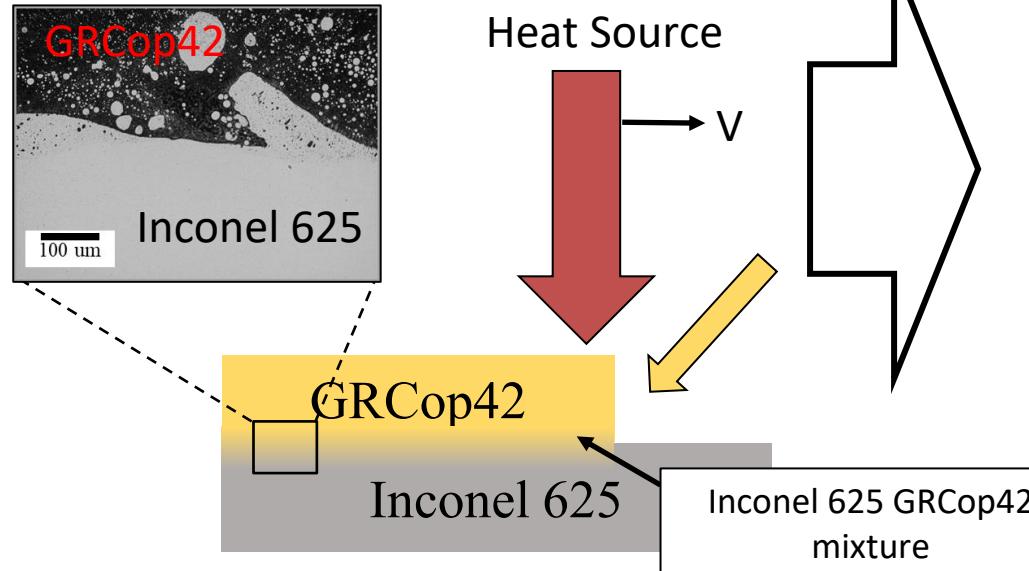
[3] Characteristics of Bi-metallic Interfaces Formed During Direct Energy Deposition Additive Manufacturing Processing

[4] Electron Beam Freeform Fabrication of Dissimilar Materials: Cracking in Inconel® 625 Deposited on GRCop-84

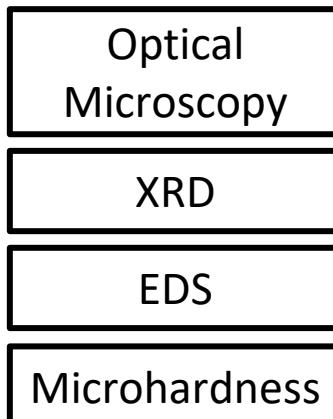
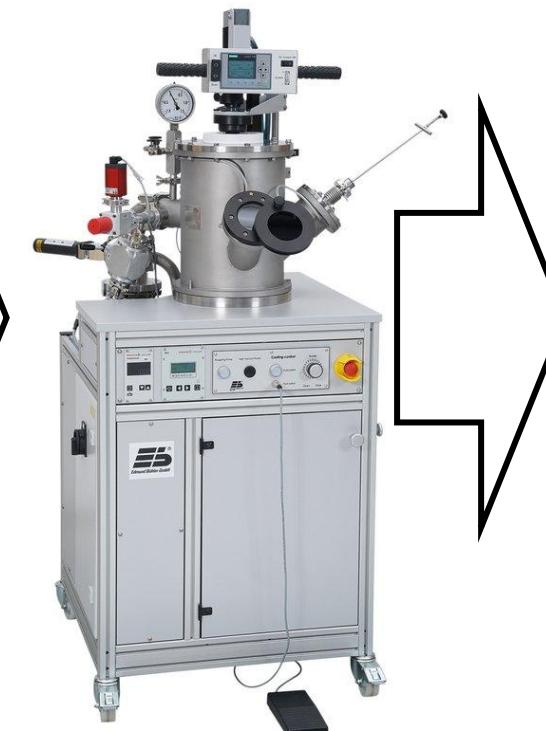
[5] Advancing GRCop-based Bimetallic Additive Manufacturing to Optimize Component Design and Applications for Liquid Rocket Engines

# Arc melting: mimicking melt pool mixing

- Arc melting is used to mimic the melt pool mixing without concerns of inhomogeneity and reheating that occurs in laser based processes.



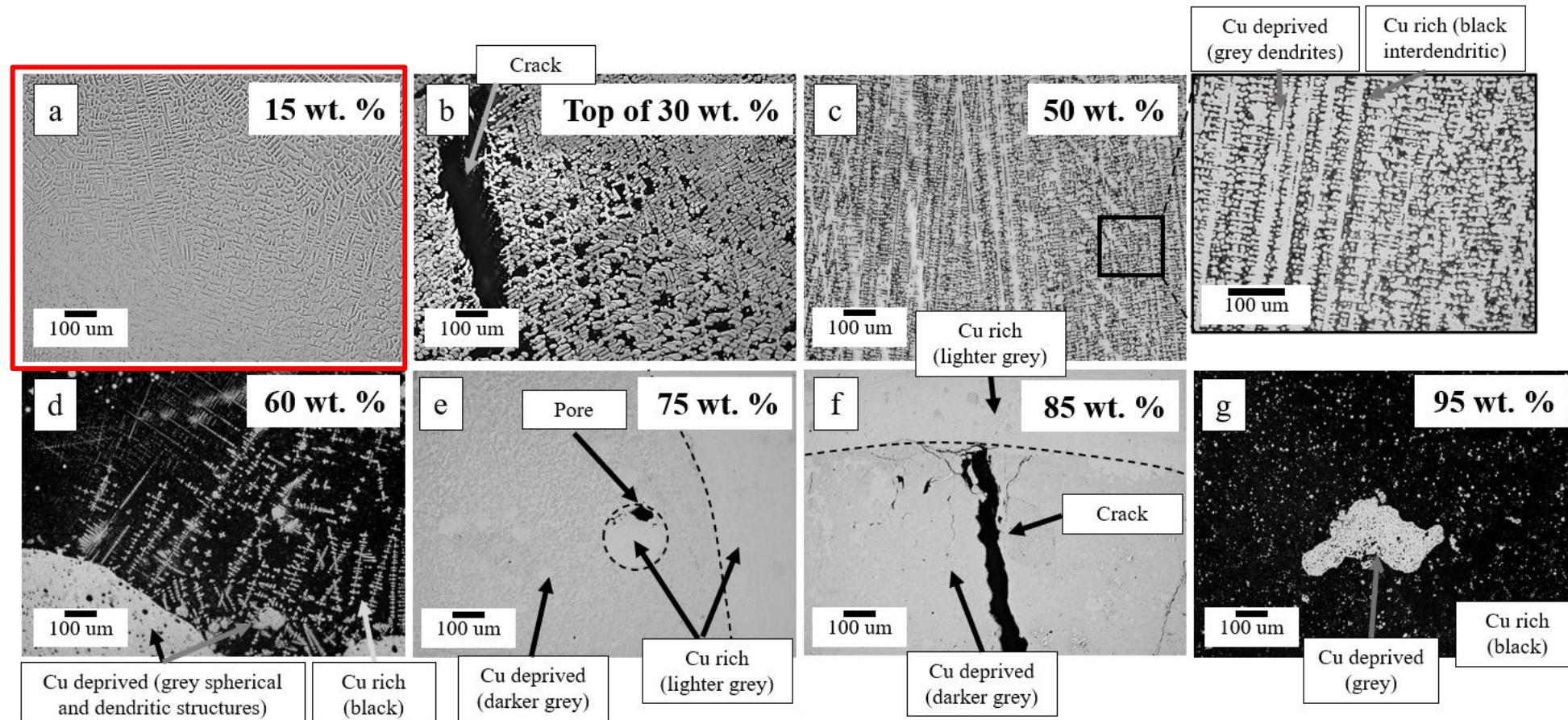
Edmund-Buhler AM200 Arc Melter



Alloy	Ni	Cr	Nb	Ta	Mo	Cu	Fe	Ti	Al	Other
Inconel 625	<b>64.8</b>	<b>22.2</b>		<b>3.49</b>	<b>8.6</b>	.016	0.26	0.21	0.14	.284
GRCop42		<b>3.3</b>	<b>2.7</b>			<i>Bal</i>				

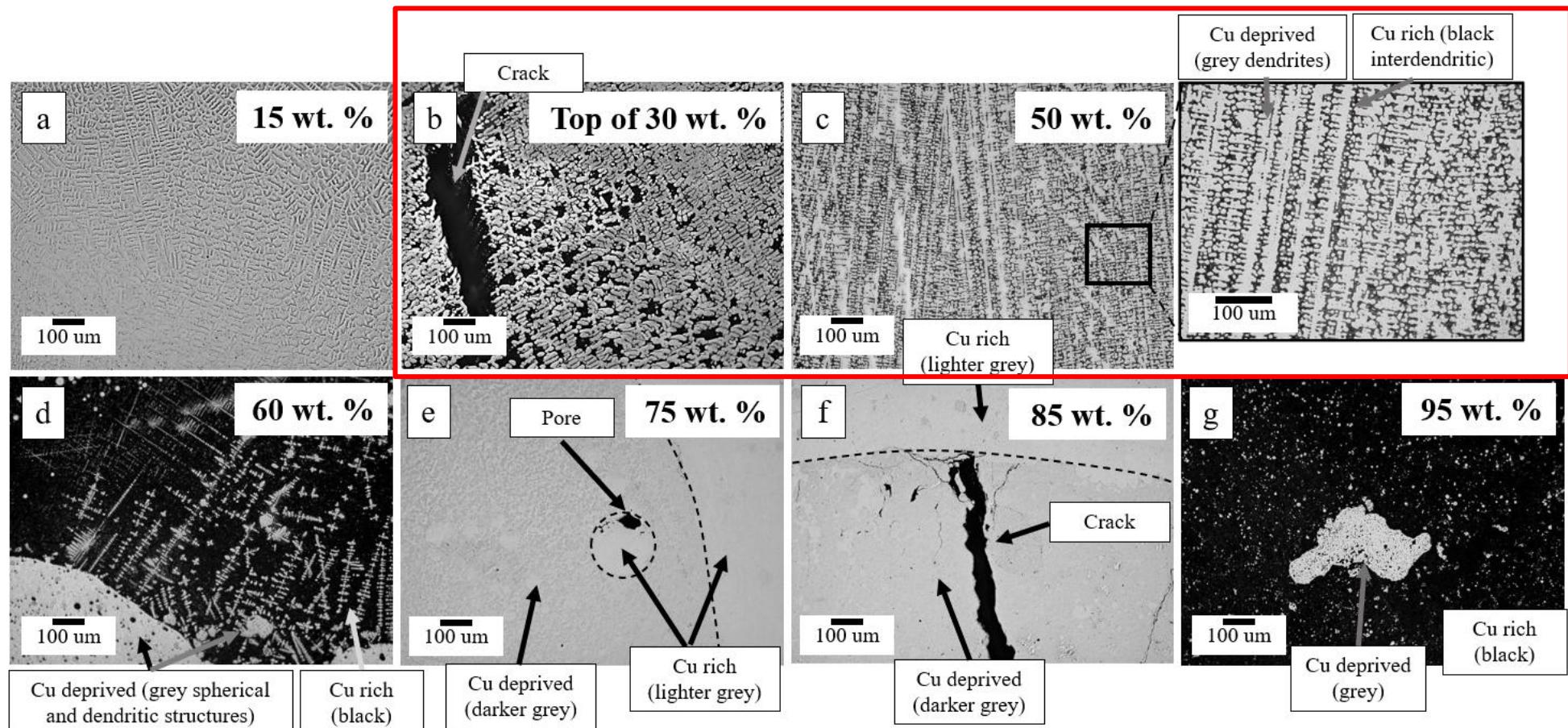
# Microstructure of Inconel 625-GRCop42 Mixtures

- Dendritic microstructure observed in 15 wt. % GRCop42 sample



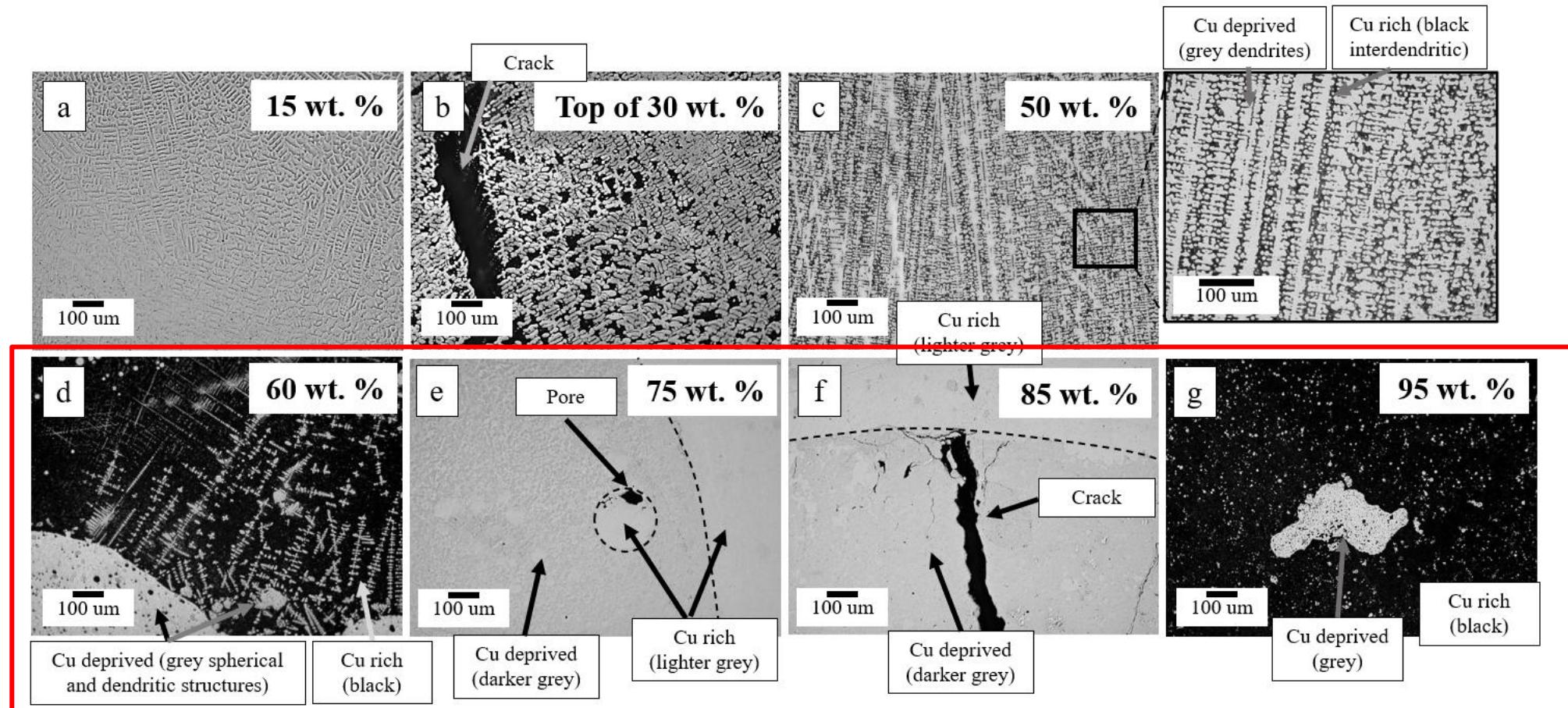
# Microstructure of Inconel 625-GRCop42 Mixtures

- 30-50 wt. % GRCop42 samples show a Cu-rich Cu-deprived dendritic morphology



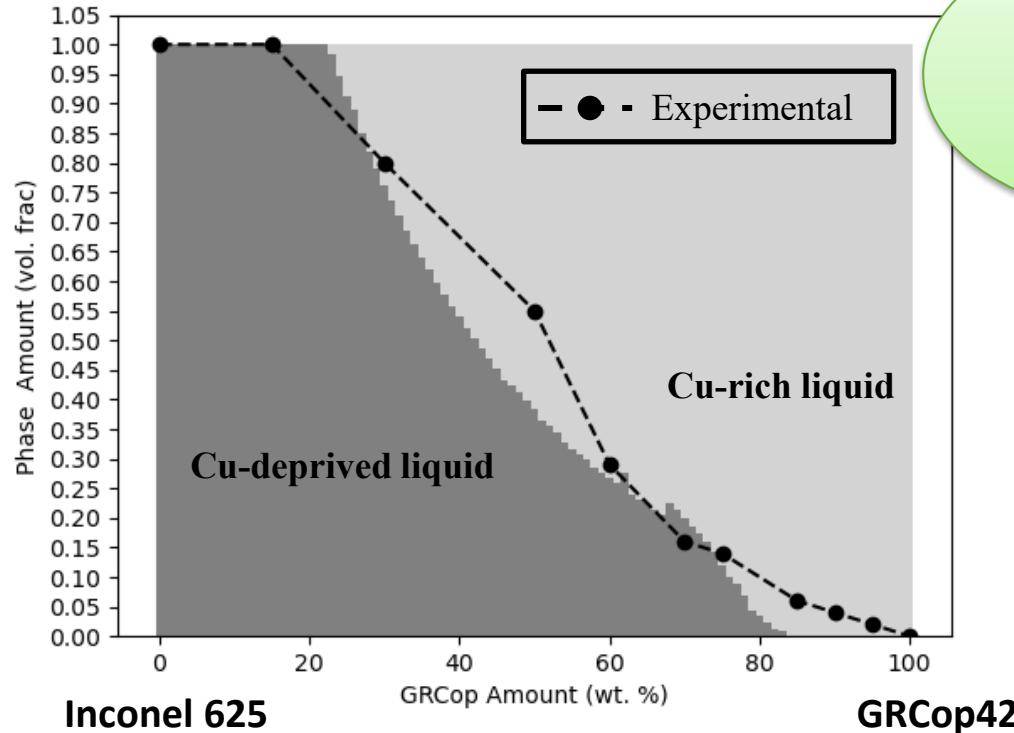
# Microstructure of Inconel 625-GRCop42 Mixtures

- 60-95 wt. % GRCop42 samples show a **Cu-deprived spherical morphology**.

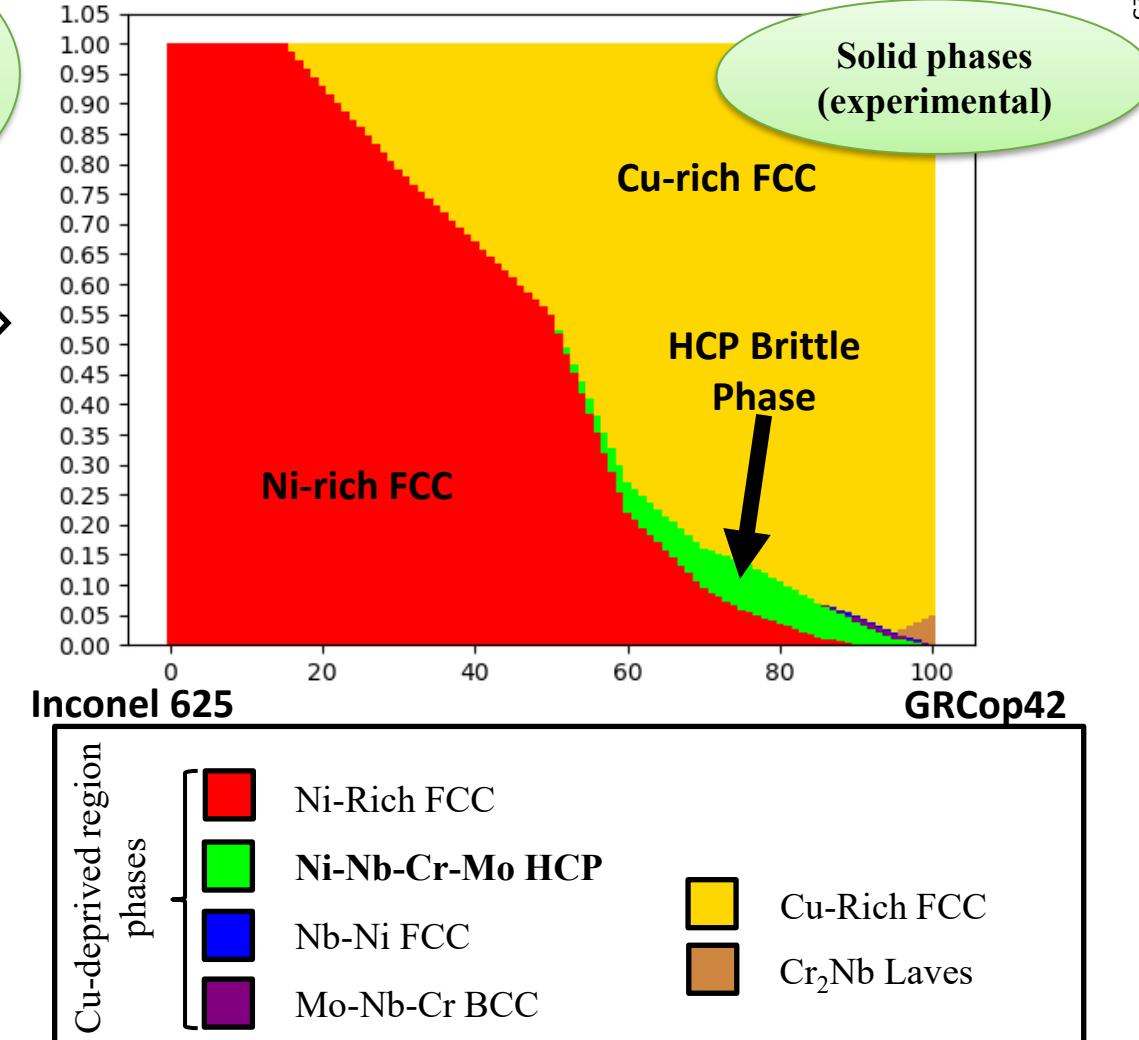


# Liquid Miscibility Gap + Phase Analysis

Liquid Miscibility gap exists in Inconel 625–GRCop42



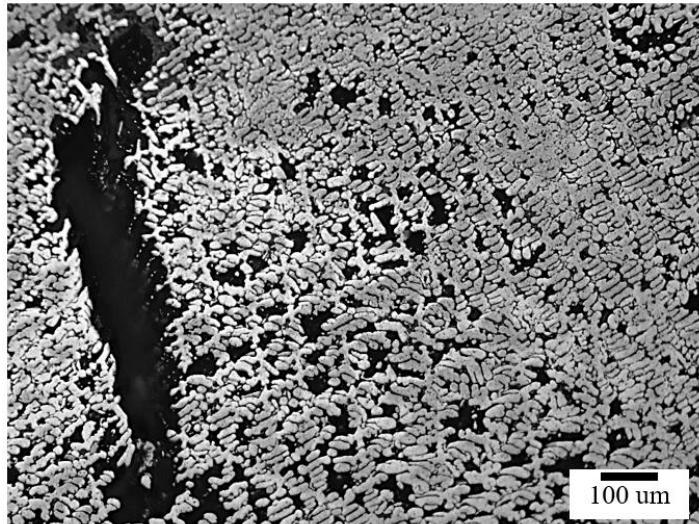
Brittle phase is formed from Cu-deprived liquid



- Miscibility gap between Cu-rich and Cu-deprived liquid exists in Inconel 625 – GRCop42 phase diagram
- This miscibility gap results in brittle phase formation from the Cu-deprived liquid at compositions greater than 50 wt. % GRCop42

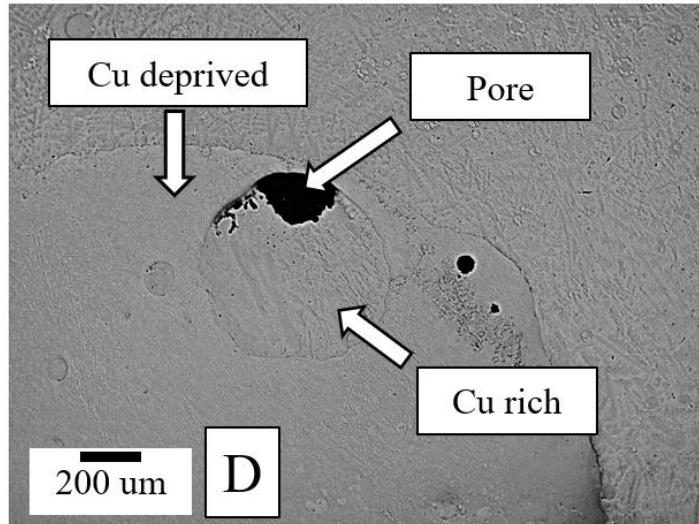
# Identifying Defects in Mixing Inconel with GRCop

- Defects in GRCop42 – Inconel 625 mixtures are found to be compositionally dependent



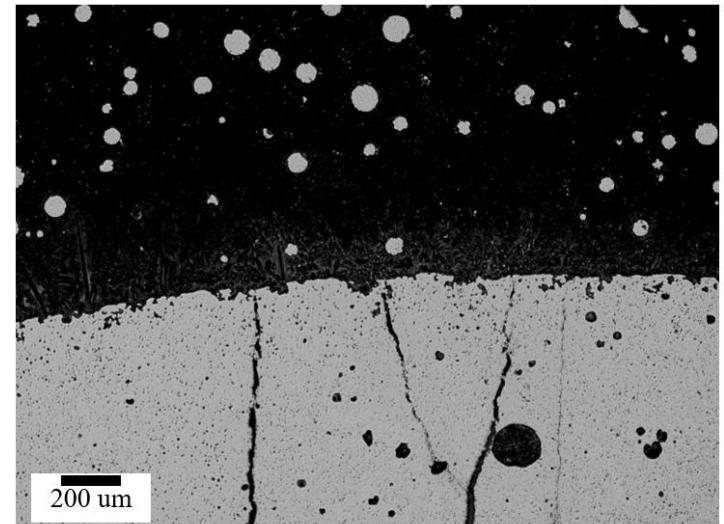
30-50 wt. % GRCop42

- Cracking observed at top 30 and 50 wt. % GRC samples
- These samples have a 2 phase dendritic microstructure



70 – 95 wt. % GRCop42

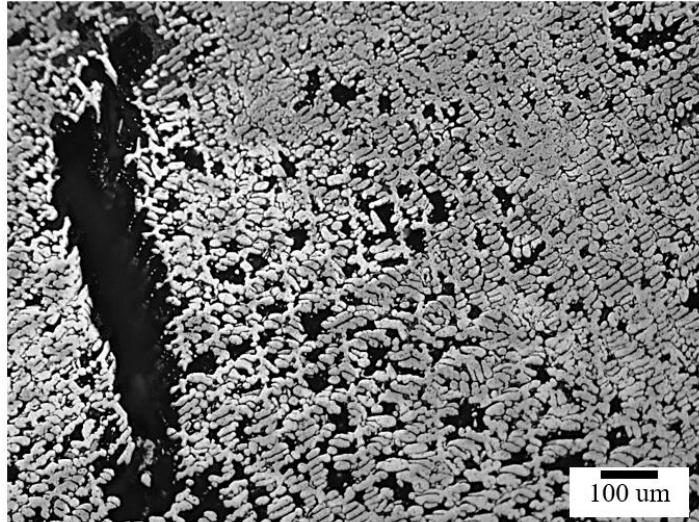
- Pores observed in the 70-95 wt. % GRC samples where Cu rich zones were trapped within Cu deprived zones.



- Cracking observed within the 70-95 wt. % Cu deprived zones.

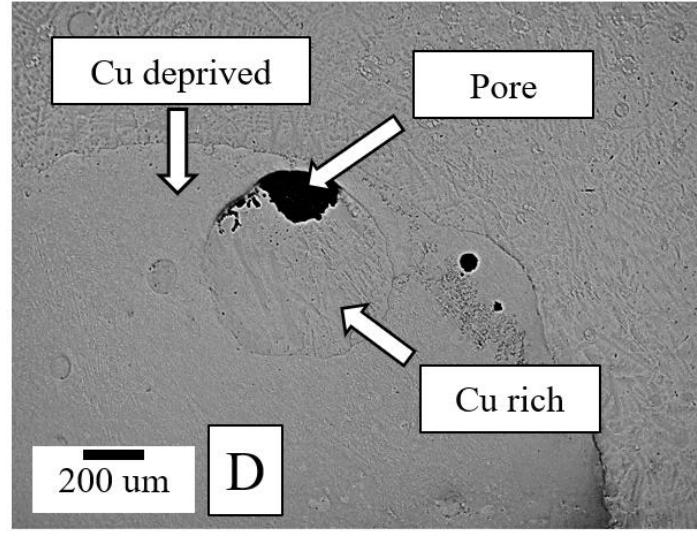
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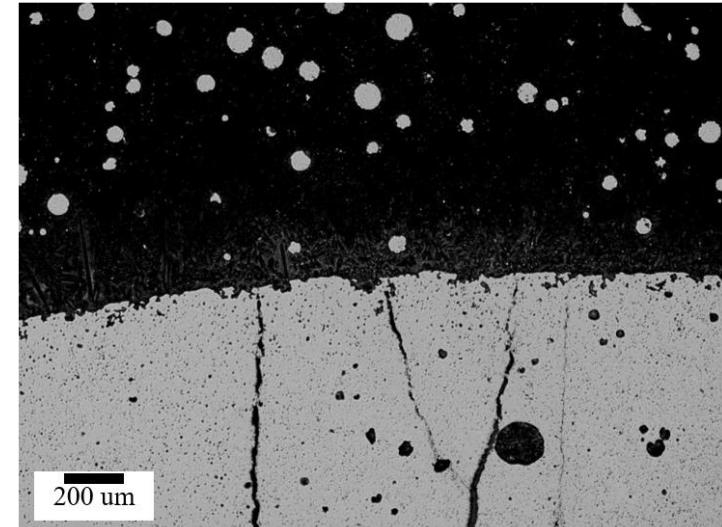
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- Cracking observed at top 30 and 50 wt. % GRC samples
- These samples have a 2 phase dendritic microstructure



60 – 95 wt. % GRCop42

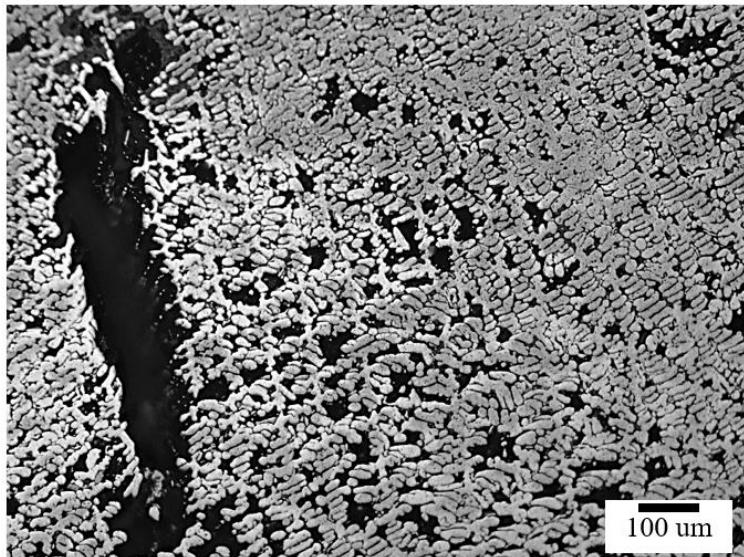
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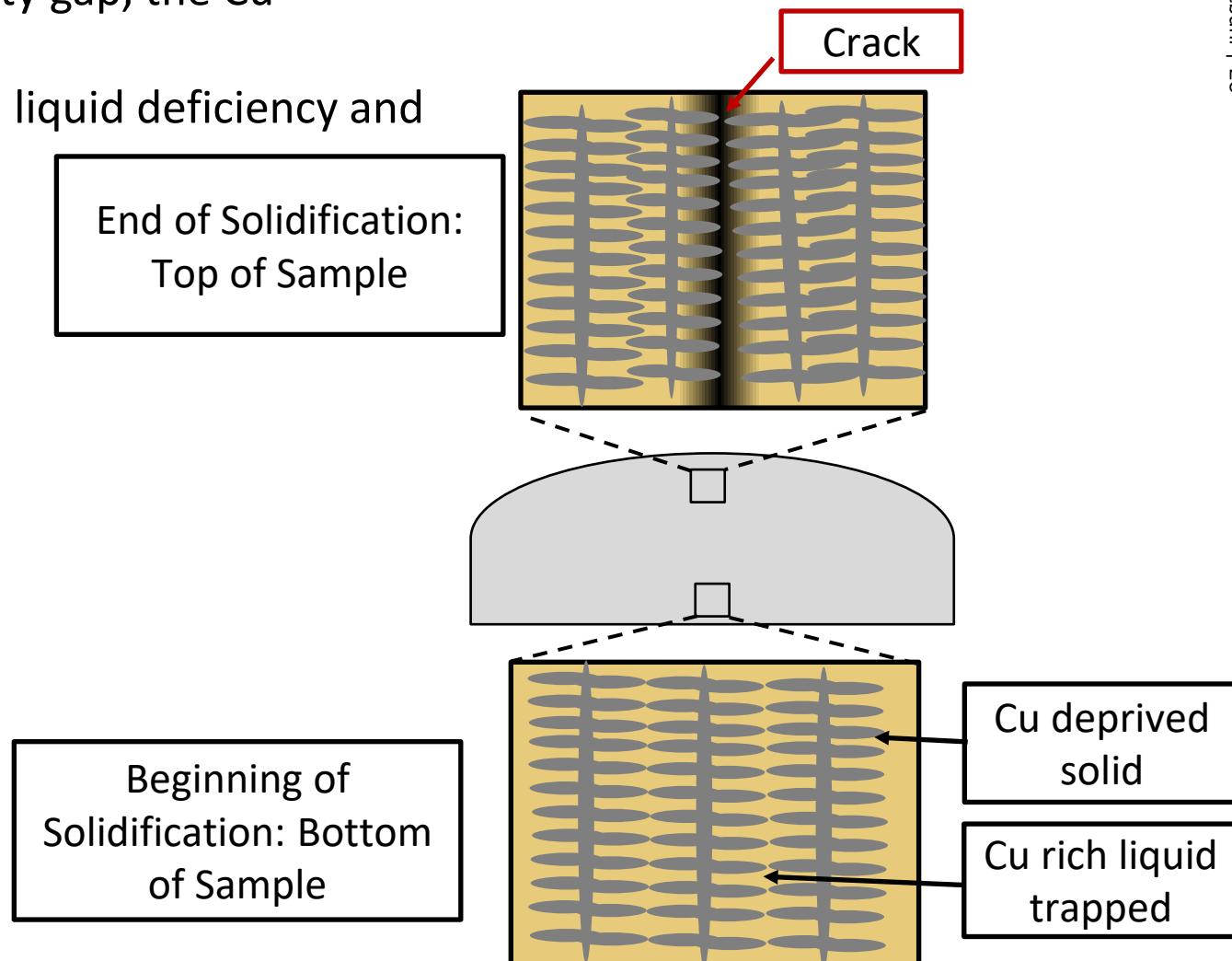
- Cracking observed within the 70-95 wt. % Cu deprived zones.

# Cracking Mechanism: Cu-Rich Liquid Entrapment

- Upon liquid separation due to the liquid miscibility gap, the Cu-deprived liquid solidifies first.
- This entraps the Cu-rich liquid, leading to Cu-rich liquid deficiency and cracking at the last point of solidification.
- This is a new cracking mechanism that has not been reported in literature.

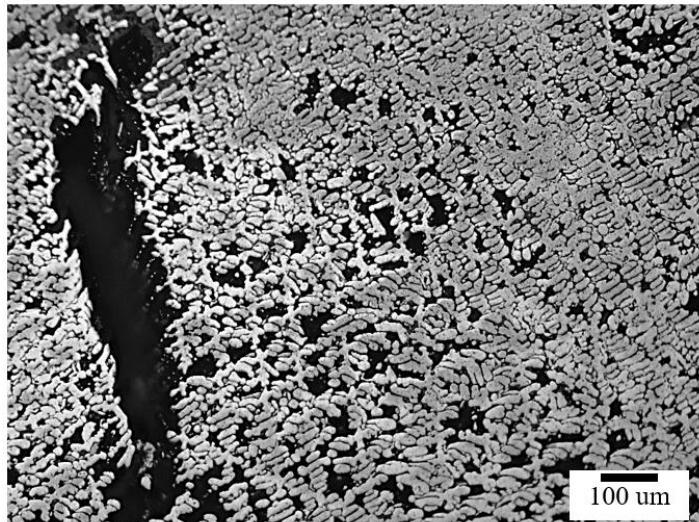


30 wt. % GRC



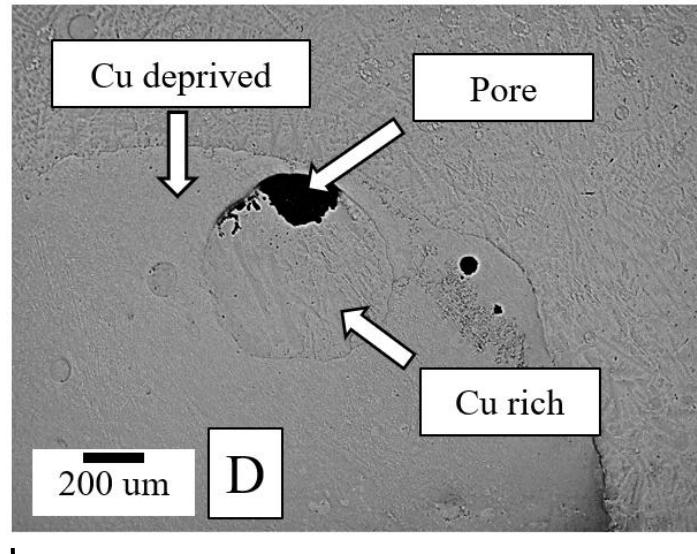
# Identifying Defects in Mixing Inconel with GRCop

- Defects in GRCop42 – Inconel 625 mixtures are found to be compositionally dependent



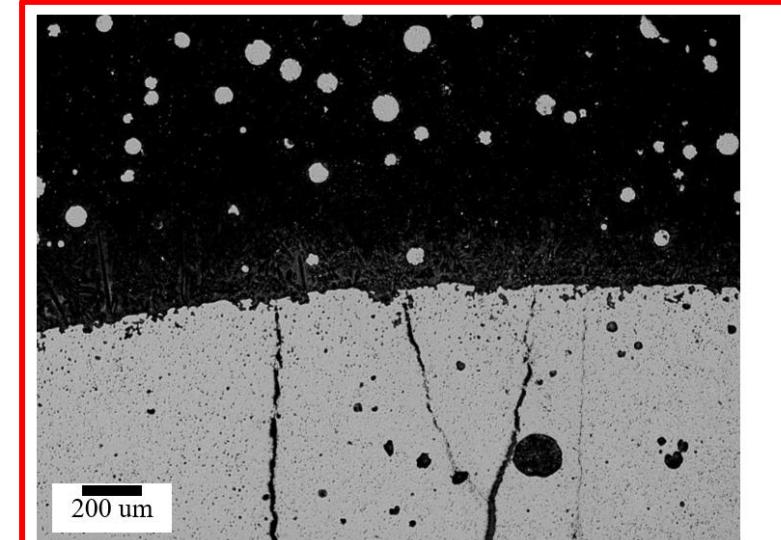
30-50 wt. % GRCop42

- Cracking observed at top 30 and 50 wt. % GRC samples
- These samples have a 2 phase dendritic microstructure



60 – 95 wt. % GRCop42

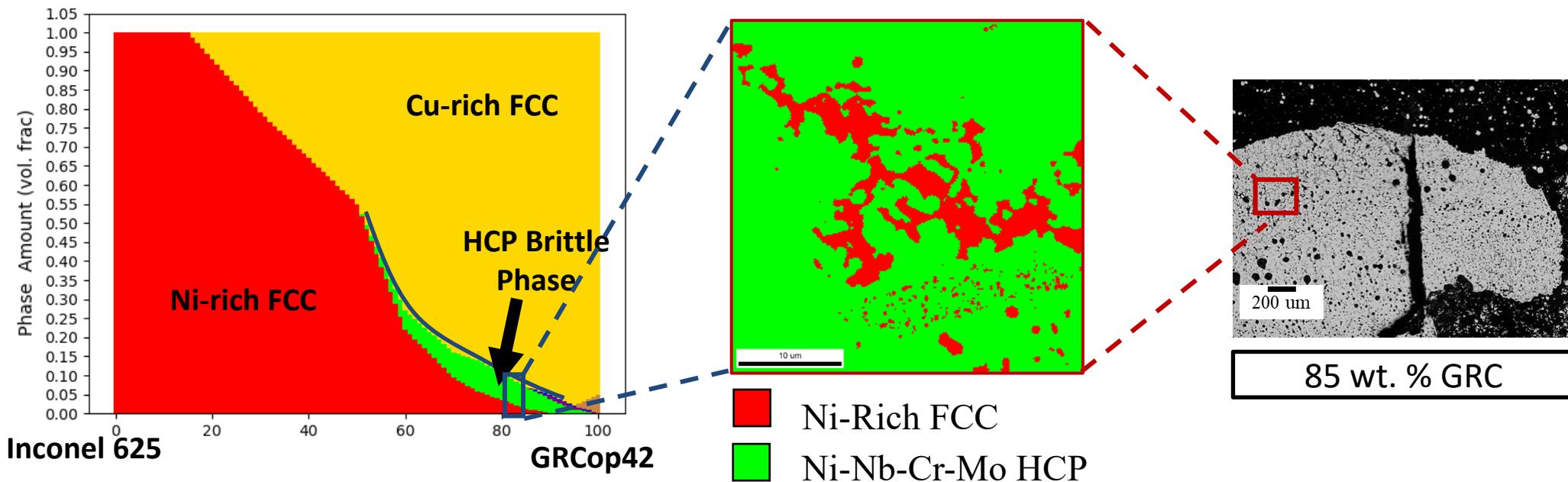
- Pores observed in the 70-95 wt. % GRC samples where Cu rich zones were trapped within Cu deprived zones.



- Cracking observed within the 70-95 wt. % Cu deprived zones.

# Cracking Mechanism: Brittle Phase + Thermal Strain

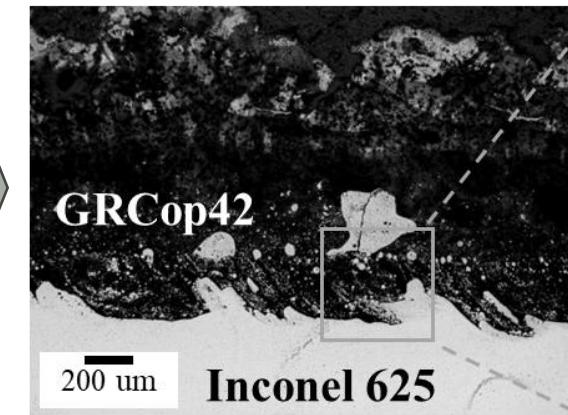
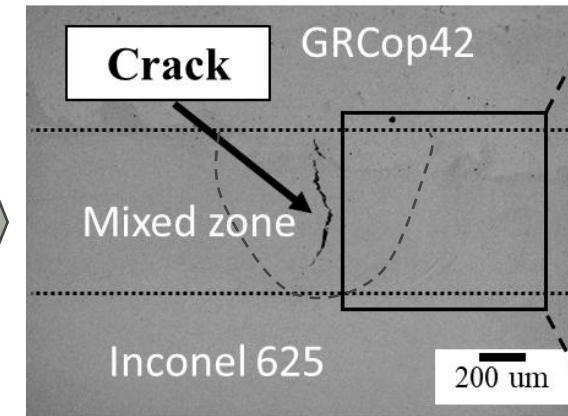
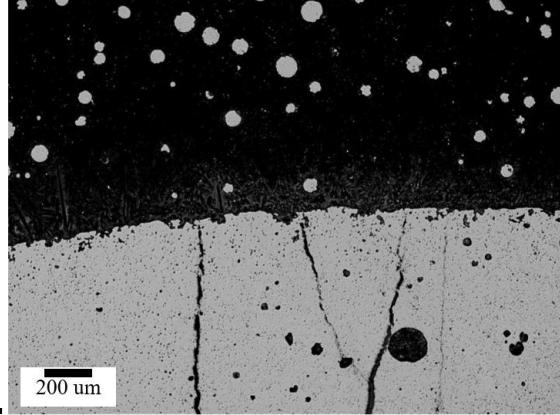
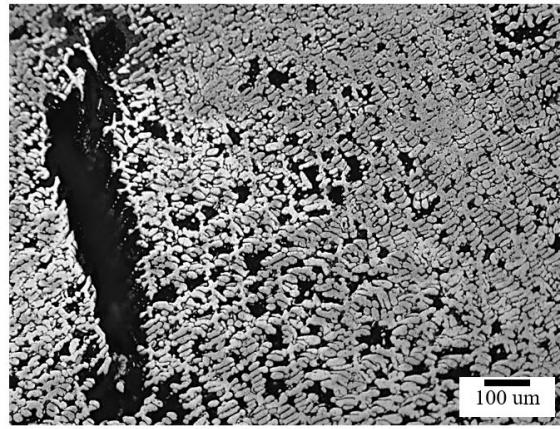
- Upon solidification and cooling, high thermal gradients occur inducing thermal strain.
- The brittle phases in the Cu-deprived region of the 60-95 wt.% GRCop samples has high brittle phase volume fraction.
- The inability of the brittle phase to resolve thermal strain results in



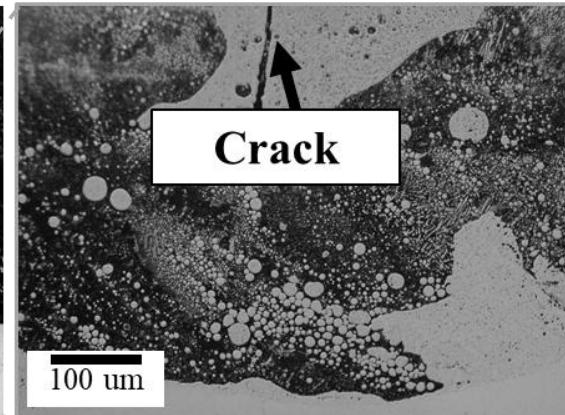
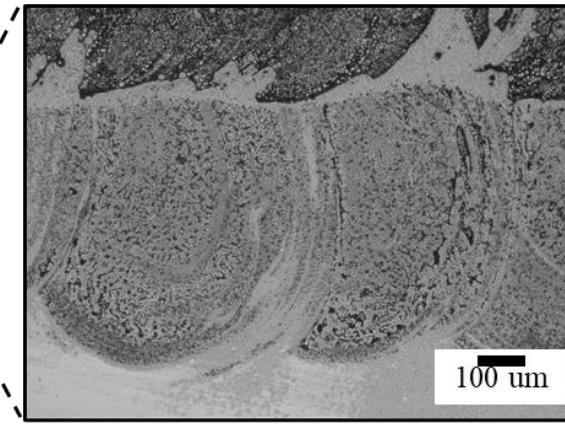
# Validation of arc-melting to find defects in dissimilar metal AM

## Arc melting

30-50 wt. %  
GRCop42



## LDED



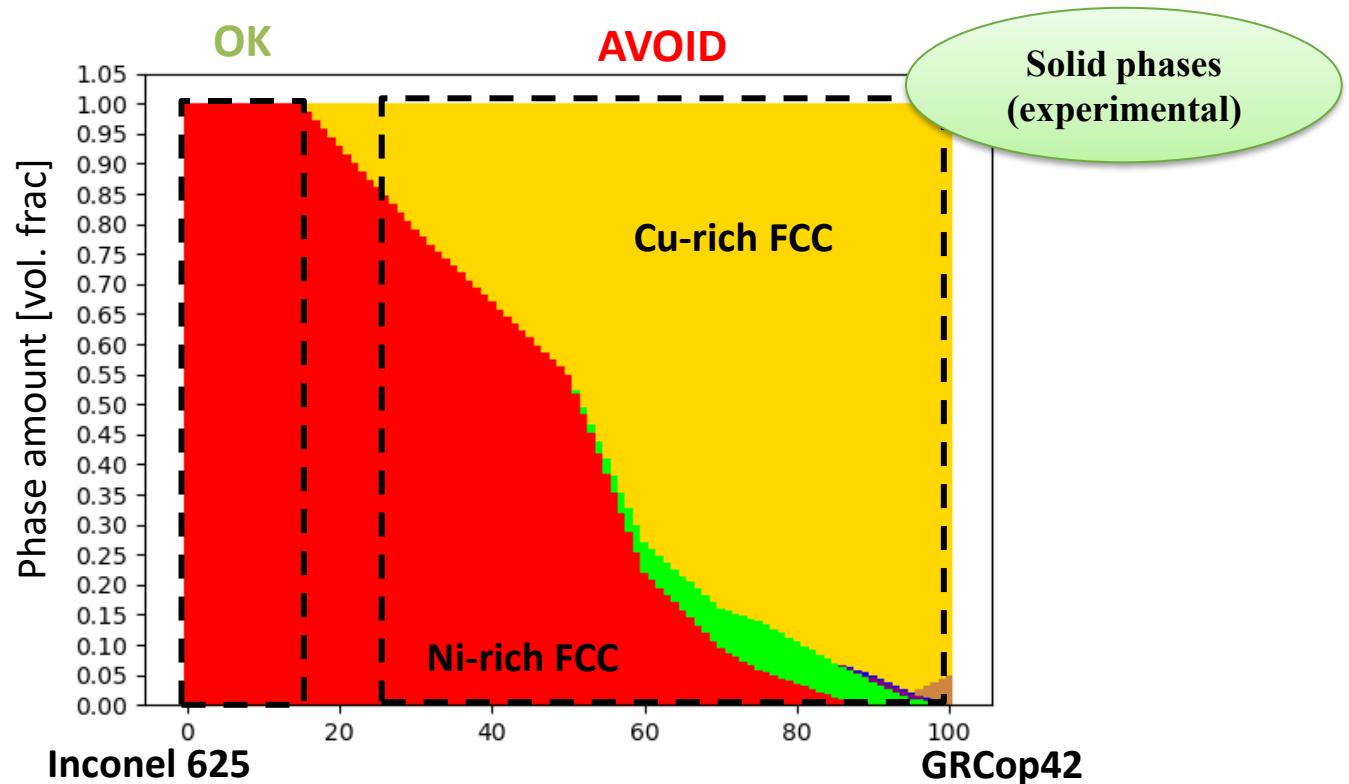
### Higher remelting LDED parameter

- More Inconel 625 substrate remelted
- Cracking observed in middle of bead
- Morphology consistent with arc melting results

### Lower remelting LDED parameter

- Less Inconel 625 substrate remelted
- Cu-deprived islands observed with cracks
- Morphology consistent with arc melting results

# Summary



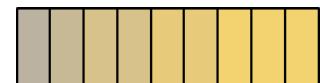
# Outline

**Goal:** Understand the process-structure-property relationship in Inconel – GRCop42 joints

- Impact of parameters for LDED processing of wire Inconel 625
  - ❖ Manuscript 1: *Preis et al., Influence of travel speed on microstructure and mechanical behavior of Inconel 625 fabricated using wire fed laser directed energy deposition, Journal of Materials Processing Technology*
- Impact of Inconel 625 GRCop42 mixing
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Inconel 625

Inconel 625 GRCop42 mixtures

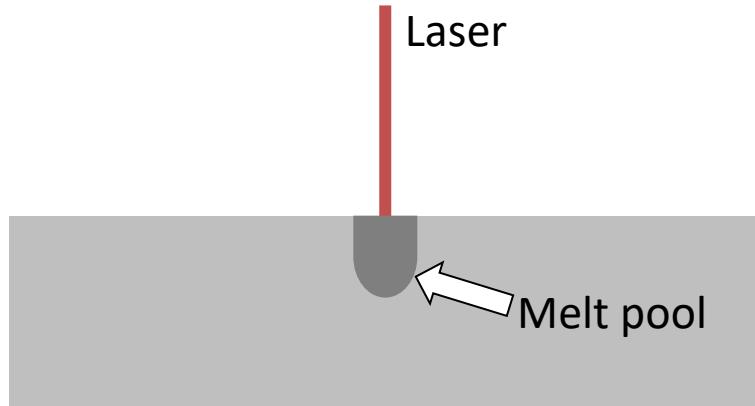


GRCop42  
Inconel 625

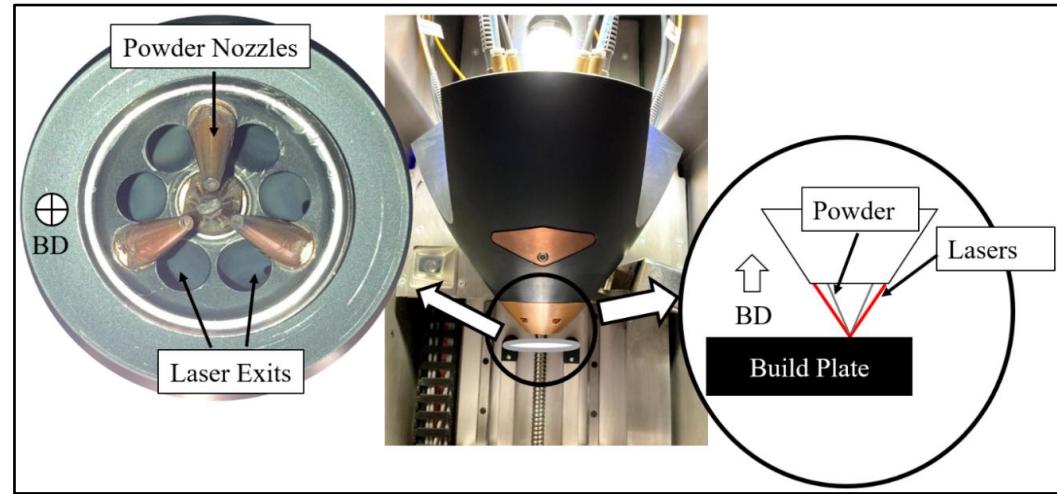
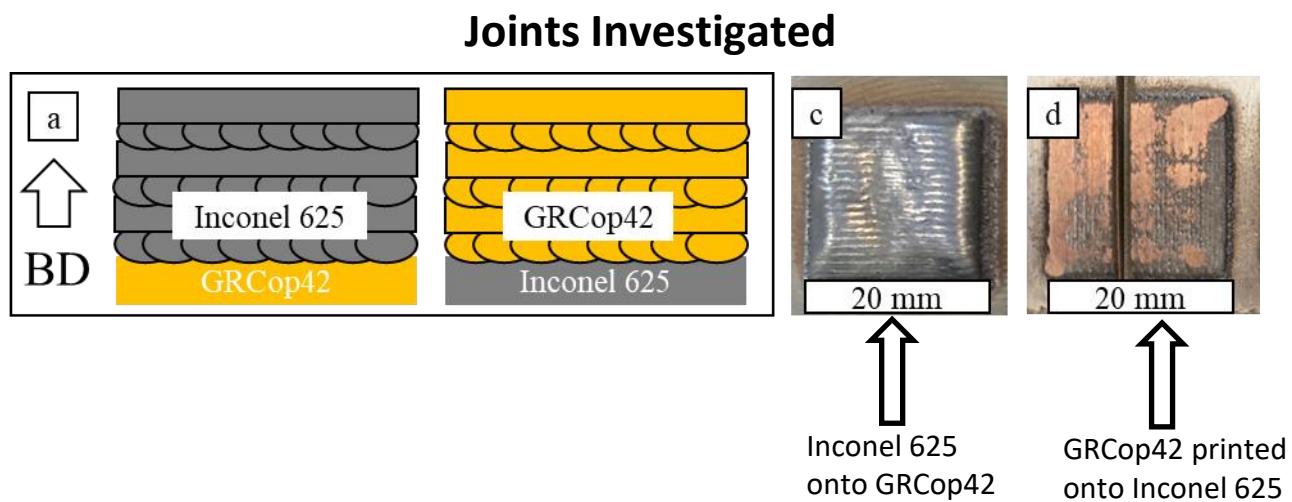
GRCop42  
Filler  
Inconel 718

## Key terms

- Melt pool – term used to describe the region of metal which is in the liquid state due to laser melting.



# Inconel GRCop Bimetallic Joint via LDED

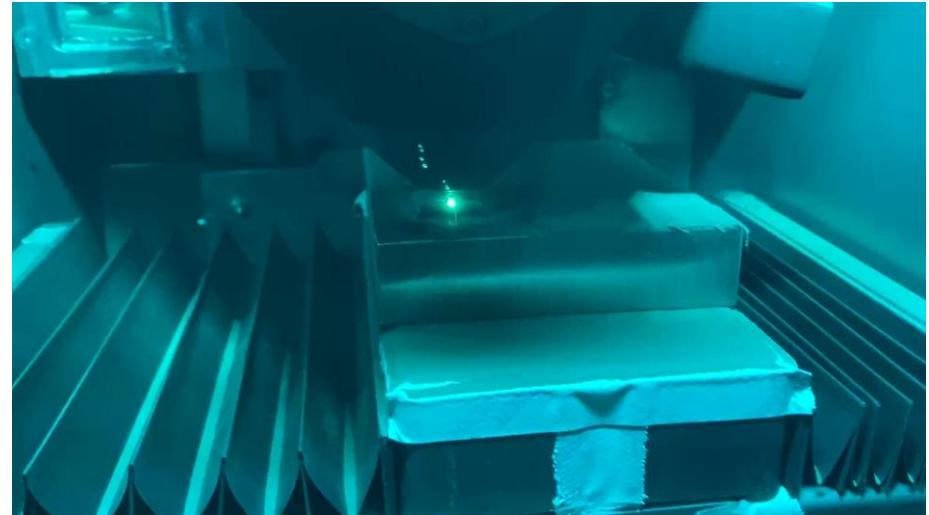
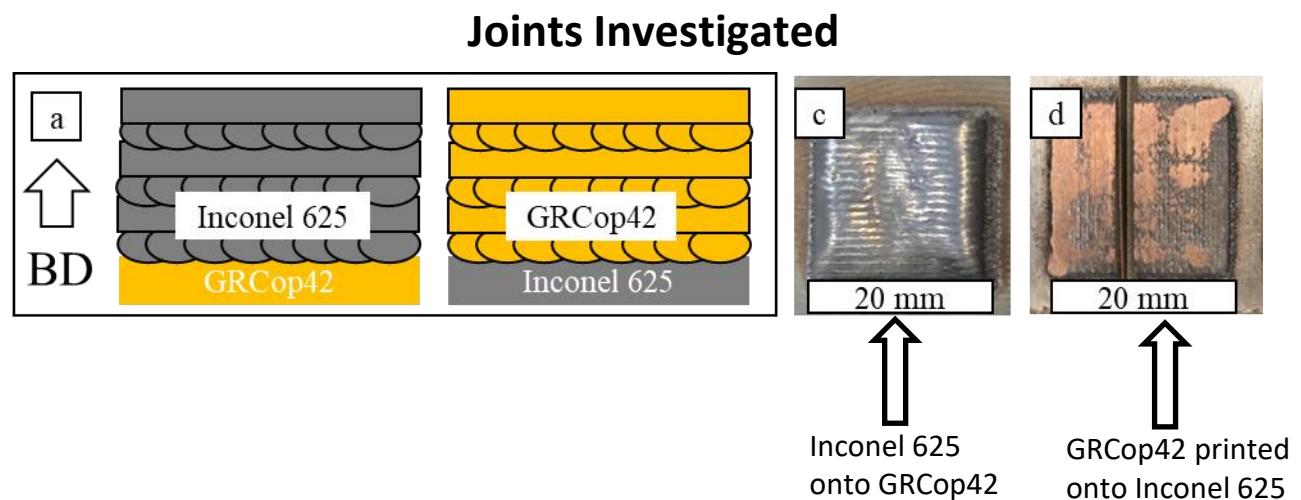


## Materials

Alloy	Ni	Cr	Nb	Ta	Mo	Cu	Fe	Ti	Al	Other
Inconel 625 Wire	64.8	22.2	3.49		8.6	.016	0.26	0.21	0.14	.284
Inconel 625 Powder	Bal	21.52	3.97		9.21		0.85	0.4	0.24	.245
GRCop42		3.3	2.7		Bal					

- Meltio M450 system used for printing. Power was varied while travel speed was kept constant at **400 mm/min.** **2 g/min** **powder** feed rate was used for all prints.
- All top alloys were deposited using **powder directed energy deposition**.

# Inconel GRCop Bimetallic Joint via LDED



## Materials

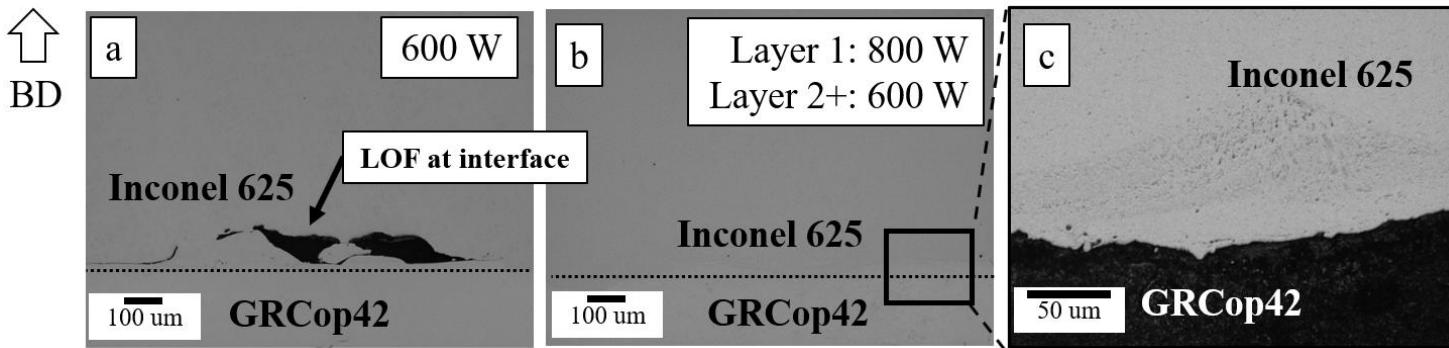
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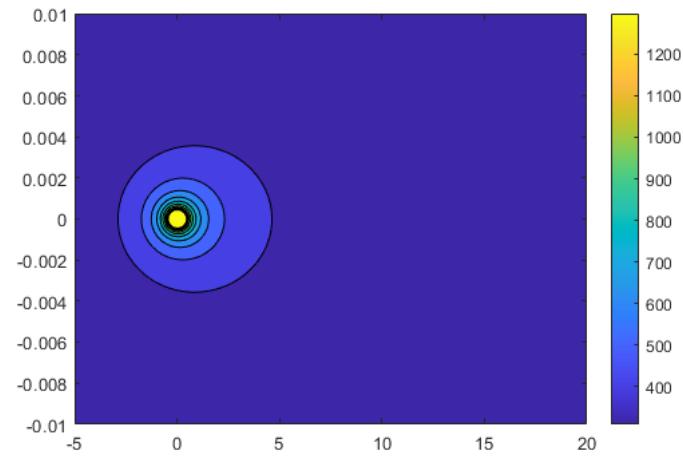
# Inconel 625 onto GRCop42



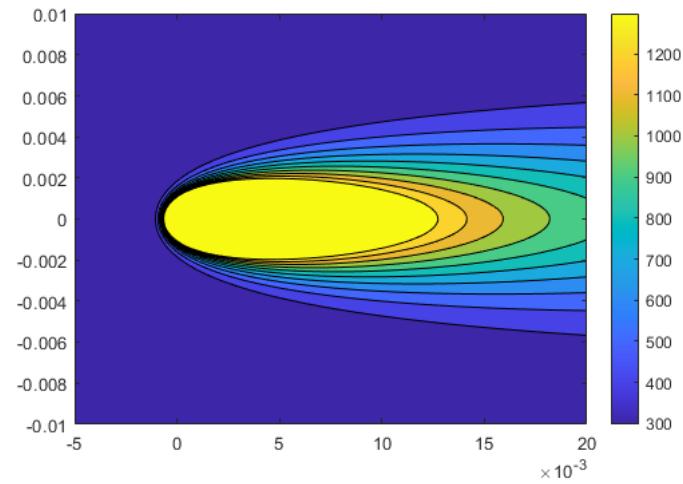
- When depositing Inconel 625 onto GRCop42 at 600 W, lack of fusion observed in the first layer due to the GRCop42 acting as a heat sink.
- The lack of fusion is avoided by increasing the laser power in the first layer.
- Defect-free joint achieved – miscibility gap avoided.



800 W, 400 mm/min onto GRCop42



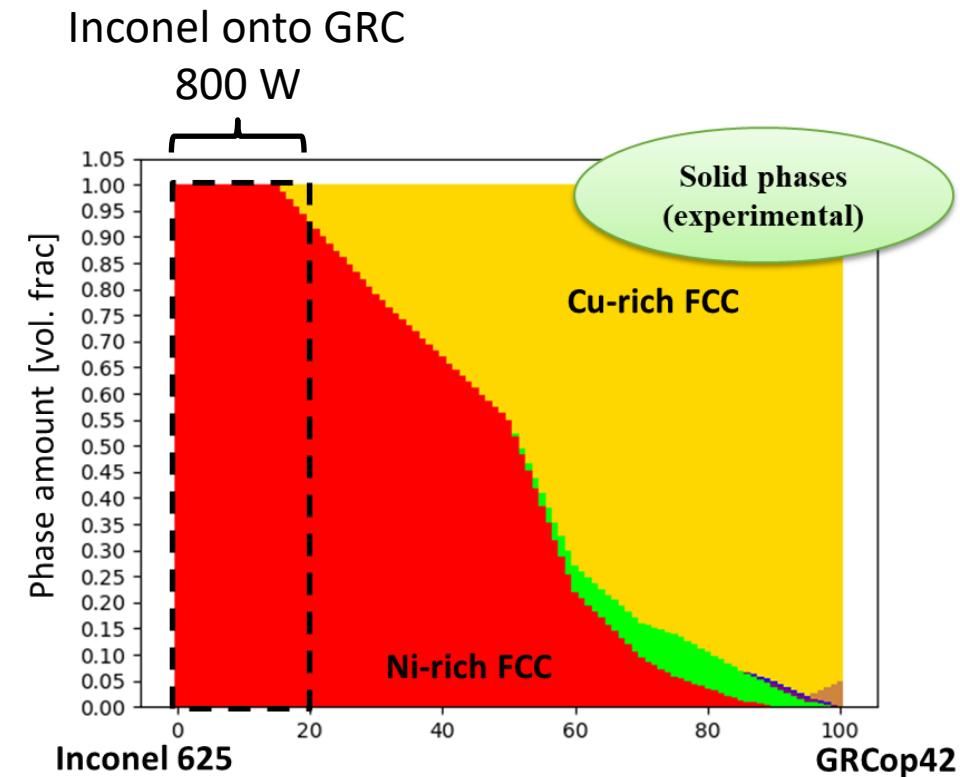
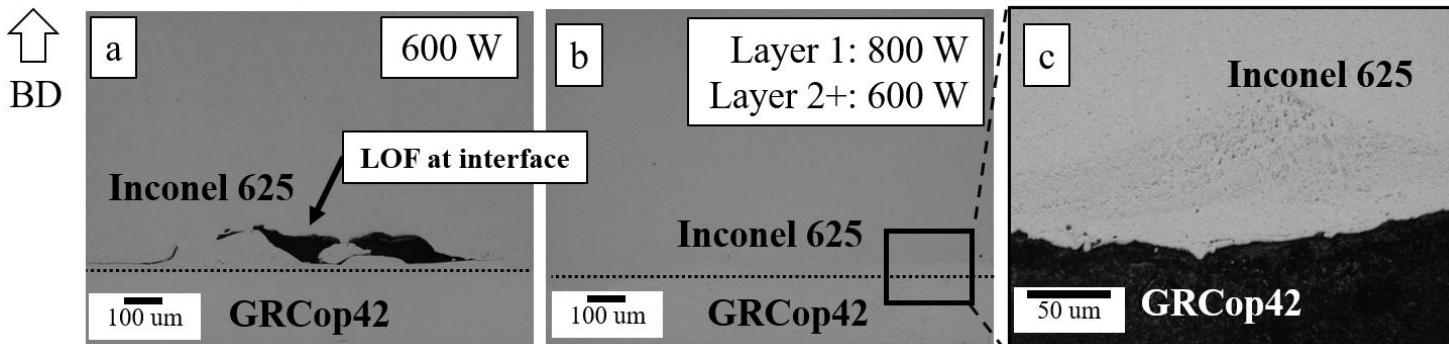
800 W, 400 mm/min onto Inconel 625



# Inconel 625 onto GRCop42

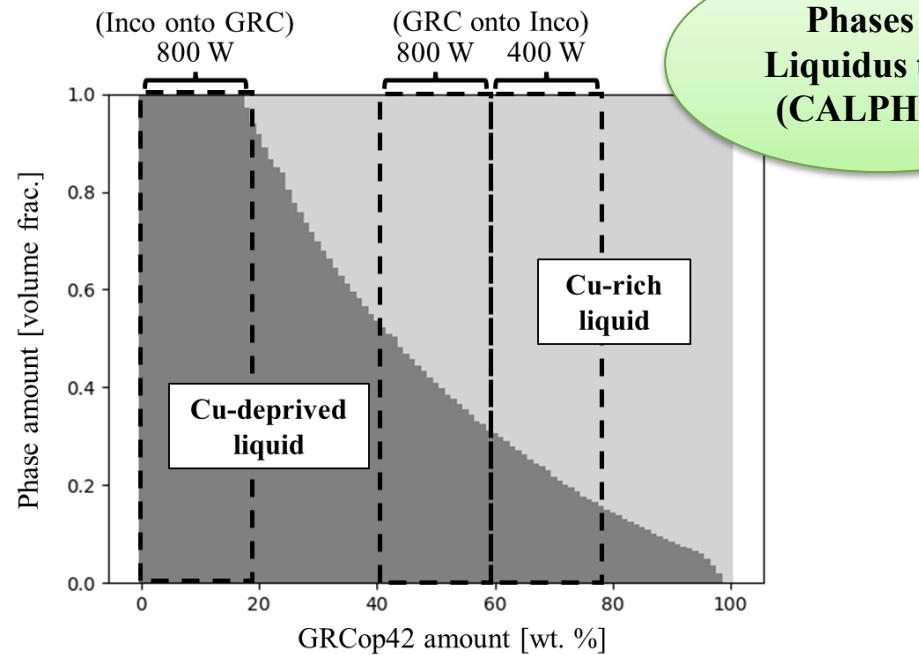


- When depositing Inconel 625 onto GRCop42 at 600 W, lack of fusion observed in the first layer due to the GRCop42 acting as a heat sink.
- The lack of fusion is avoided by increasing the laser power in the first layer.
- Defect-free joint achieved – miscibility gap avoided by remelting less than 20 wt.% GRCop42.

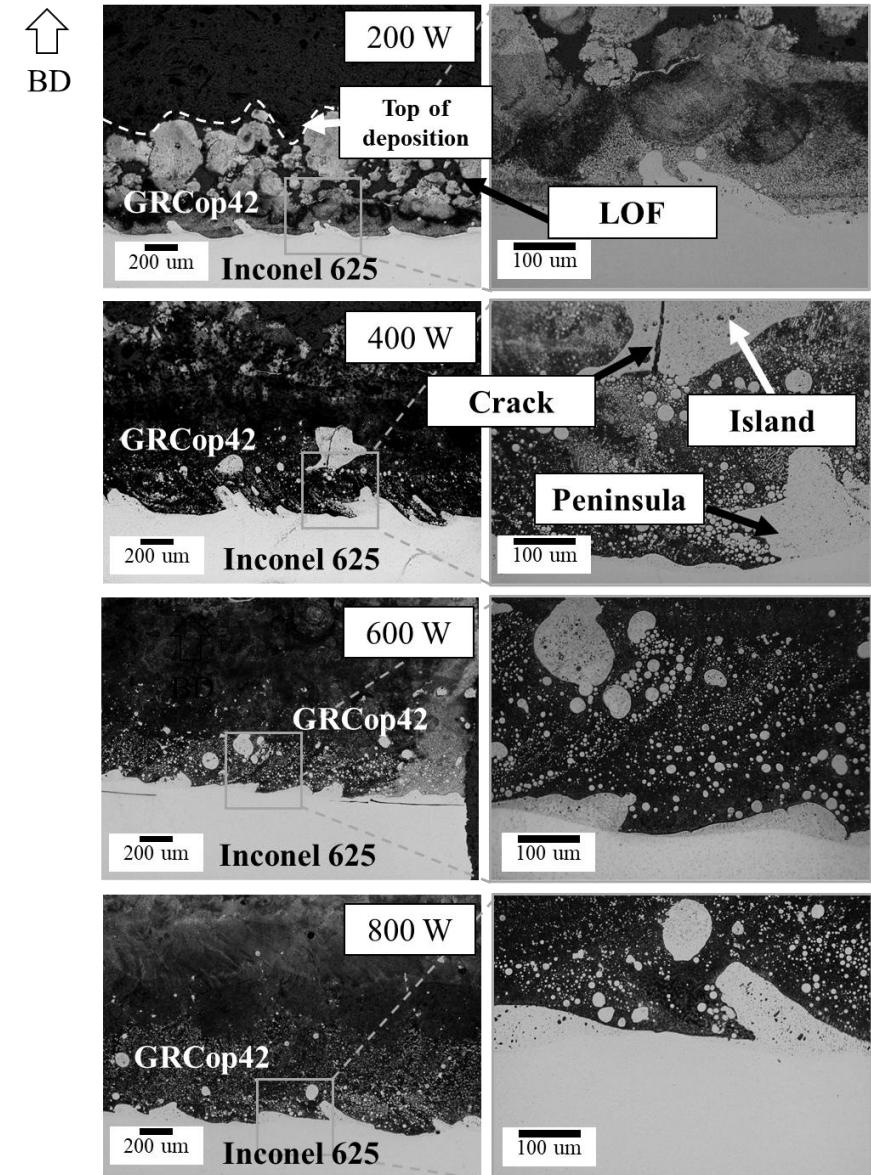


# GRCop42 onto Inconel 625

- **Islands** and **peninsulas** observed above the Inconel 625 GRCop42 interface.
- Presence of **islands** and **peninsulas** can be explained by presence of liquid miscibility gap.
- **Cracks** present in islands of 400 W sample.



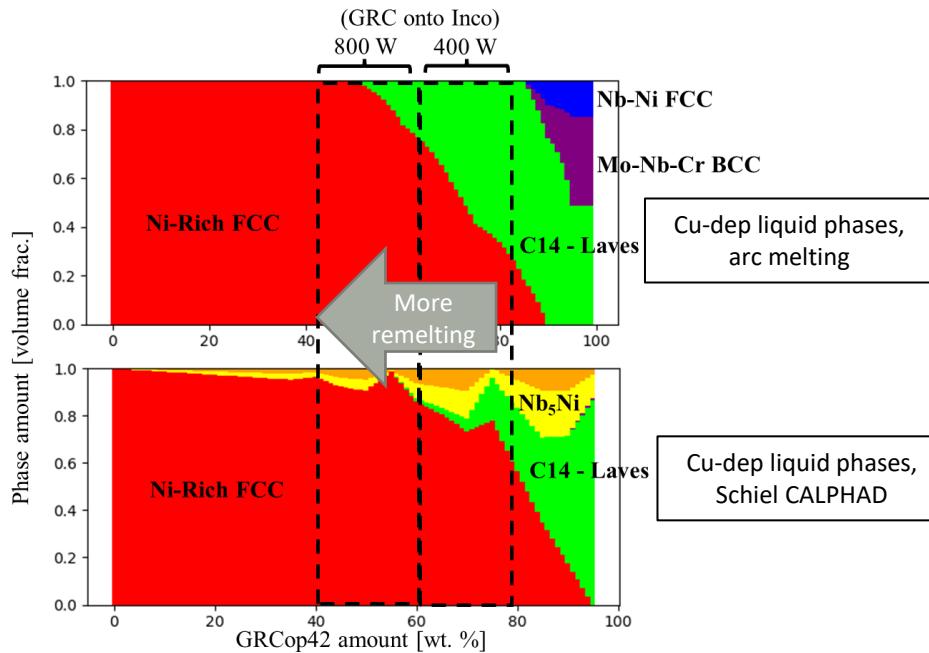
GRCop42 Powder  
Inconel 625



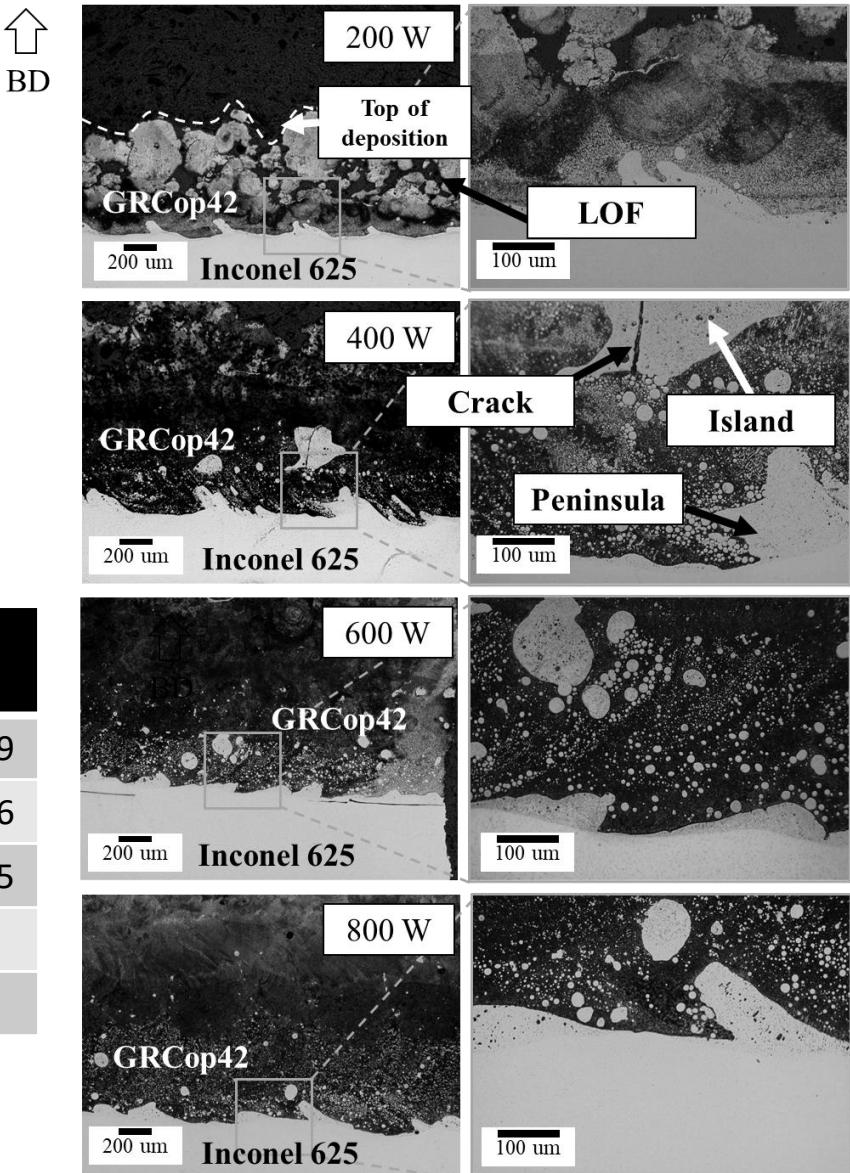
# GRCop42 onto Inconel 625

- High microhardness Cu-deprived islands and peninsulas observed.
- Cracks present in islands of 400 W sample.
- By increasing laser power, the amount of remelting of GRCop42 with Inconel 625 was increased, creating islands with a lower hardness / lower brittle phase volume fraction.

**Increased power results in fewer brittle phases, but does not eliminate them**



Sample	Peninsula HV.5	Island HV.5
400 W	$382 \pm 75$	$718 \pm 159$
600 W	$400 \pm 105$	$665 \pm 216$
800 W	$401 \pm 134$	$590 \pm 215$
GRCop42		$114 \pm 28$
Inconel 625		$262 \pm 27$



# Summary

Inconel 625 Powder  
GRCop42

GRCop42 Powder  
Inconel 625

**NO BRITTLE PHASE FORMATION / CRACKING**

**CRACKING IN ISLANDS DUE TO BRITTLE PHASES**

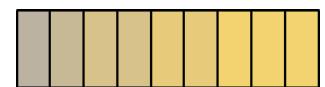
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Inconel 625

Inconel 625 GRCop42 mixtures

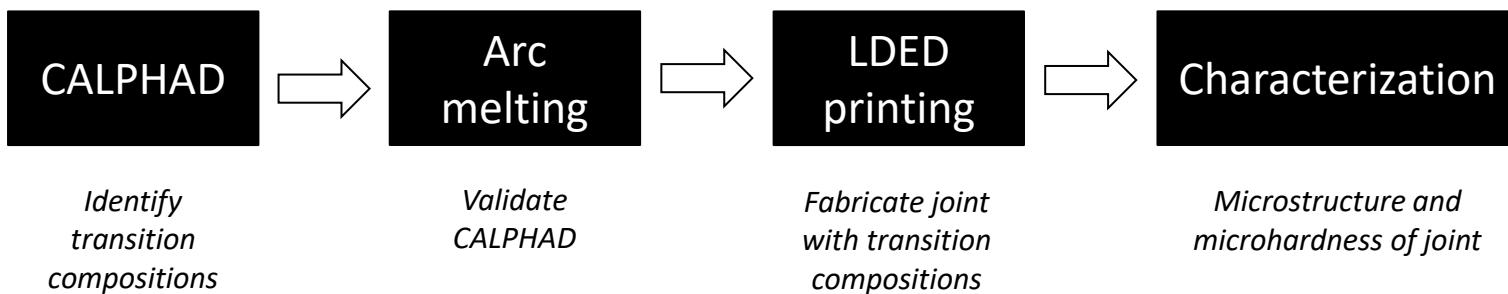


GRCop42  
Inconel 625

GRCop42  
Filler  
Inconel 718

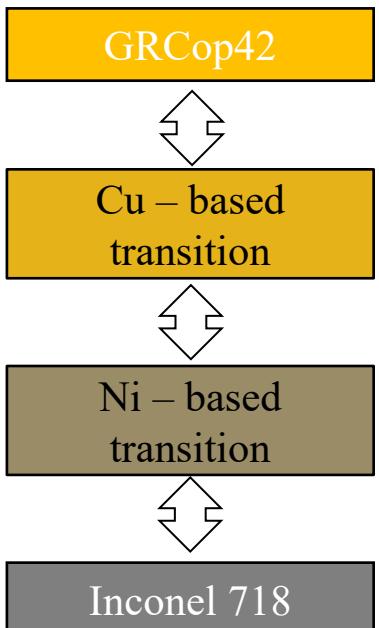
# Developing transition compositions

- Inconel 718 is a Ni-based superalloy similar to Inconel 625.
- Goal: develop and fabricate a GRCop42 Inconel 718 joint which avoids the brittle phase formation by use of developed transition compositions.



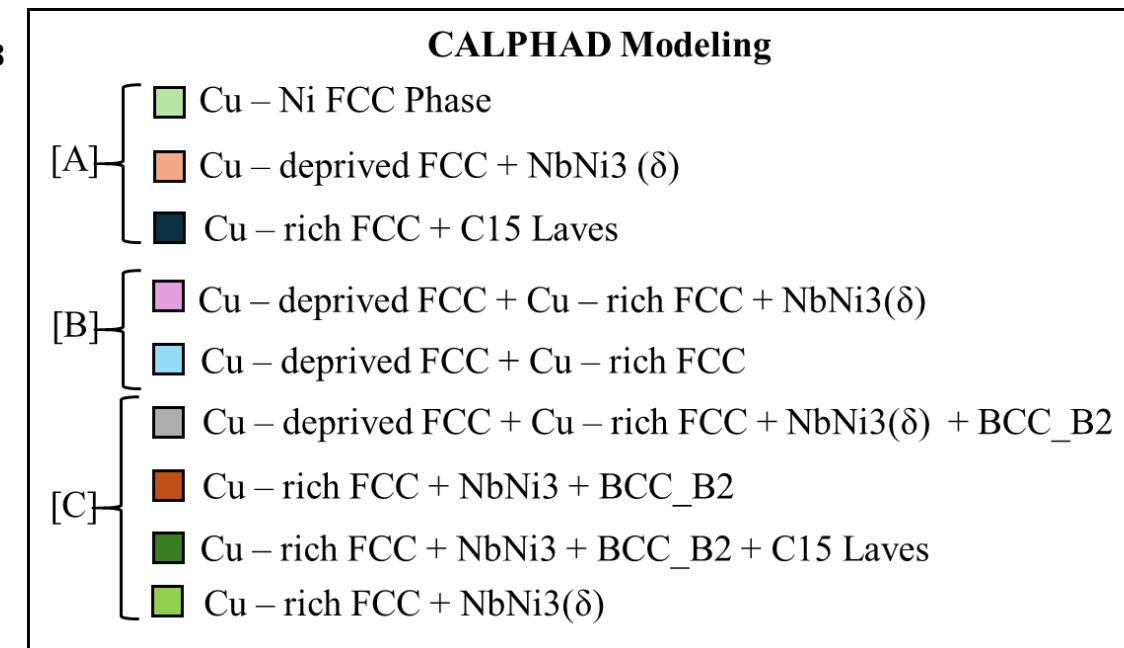
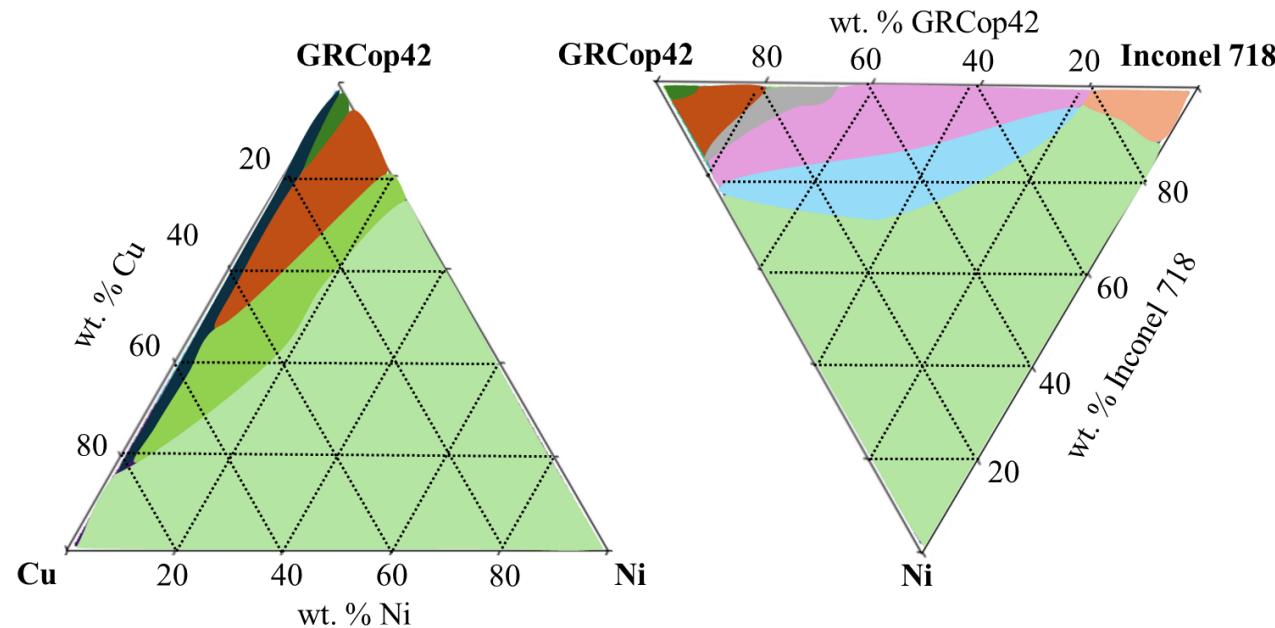
Material	Use	Ni	Cr	Nb	Mo	Cu	Fe	Al	Other
Inconel 718 Wire	AM, P	52.3	17.94	5.1	2.96	-	18.74	0.49	-
Ni Wire	AM	99.6	-	-	-	<0.01	0.17	-	0.2
Cu Wire	AM	-	-	-	-	99.9	-	-	0.1
Ni Powder	P	99.9	-	-	-	<0.01	<0.01	-	-
Cu Powder	P	-	-	-	-	99.95	-	-	0.05
GRCop42 Powder	AM,P	-	3.3	2.7	-	Bal	-	-	-

*AM for materials used in arc melting, P for materials used for printing*



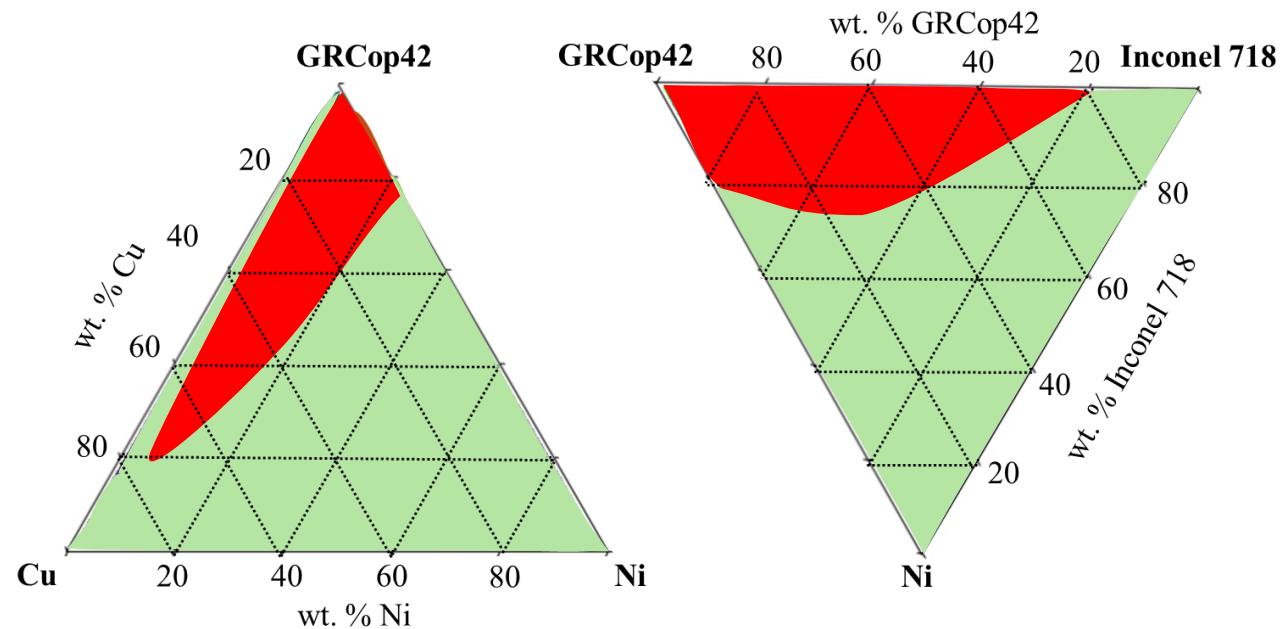
# CALPHAD Predictions

- Methodology: Use Cu and Ni based filler composition.
  - Cu is fully soluble with GRCop42
  - Ni if fully soluble with Cu
  - Ni is fully soluble with Inconel 718.
- Brittle phases predicted at regions with high wt. % of GRCop42.



# CALPHAD Predictions

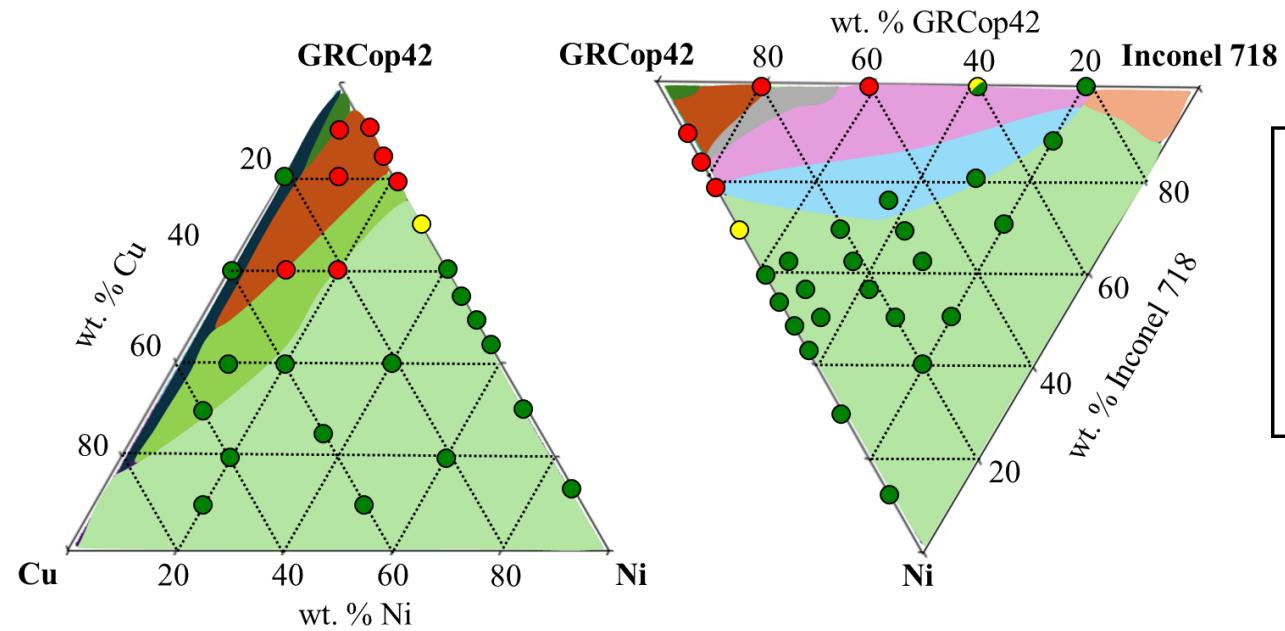
- Methodology: Use Cu and Ni based filler composition.
  - Cu is fully soluble with GRCop42
  - Ni if fully soluble with Cu
  - Ni is fully soluble with Inconel 718.
- Brittle phases predicted at regions with high wt. % of GRCop42.



█ Brittle phase formation + liquid miscibility gap (avoid)  
█ No brittle phase formation and liquid miscibility gap

# Arc melted samples

- Arc melted samples confirm “no go” region for joining Inconel 718 and GRCop42.



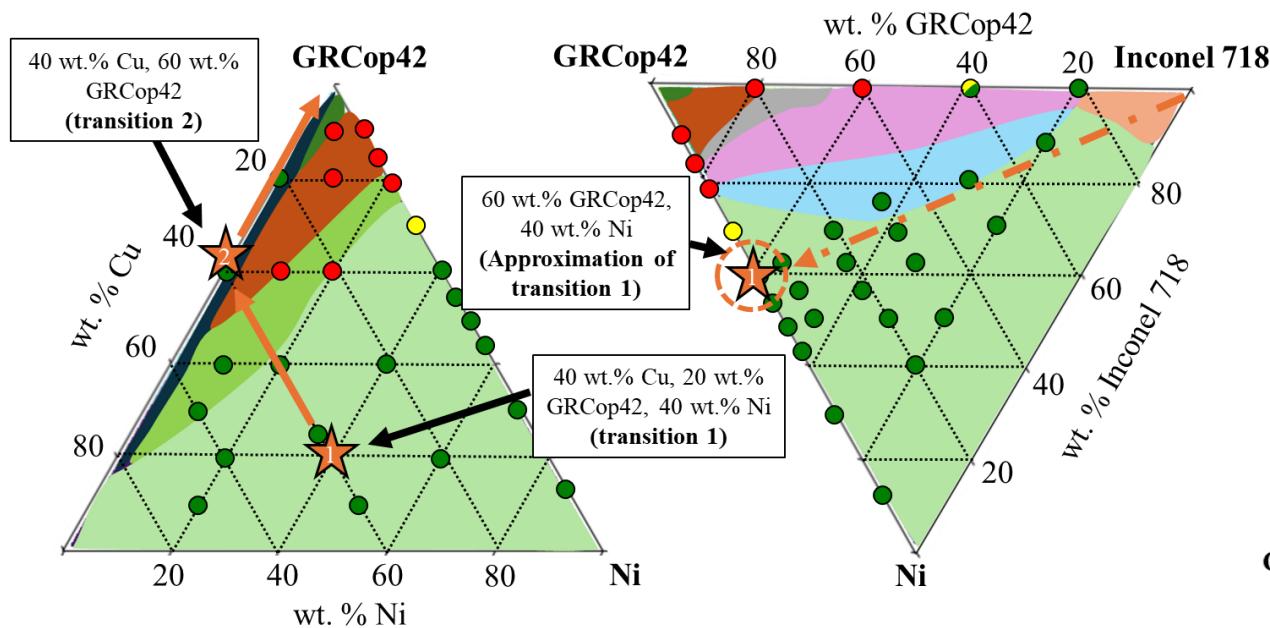
## Experimental results

- No cracking / brittle phases
- Cracking
- Brittle phases (islands)

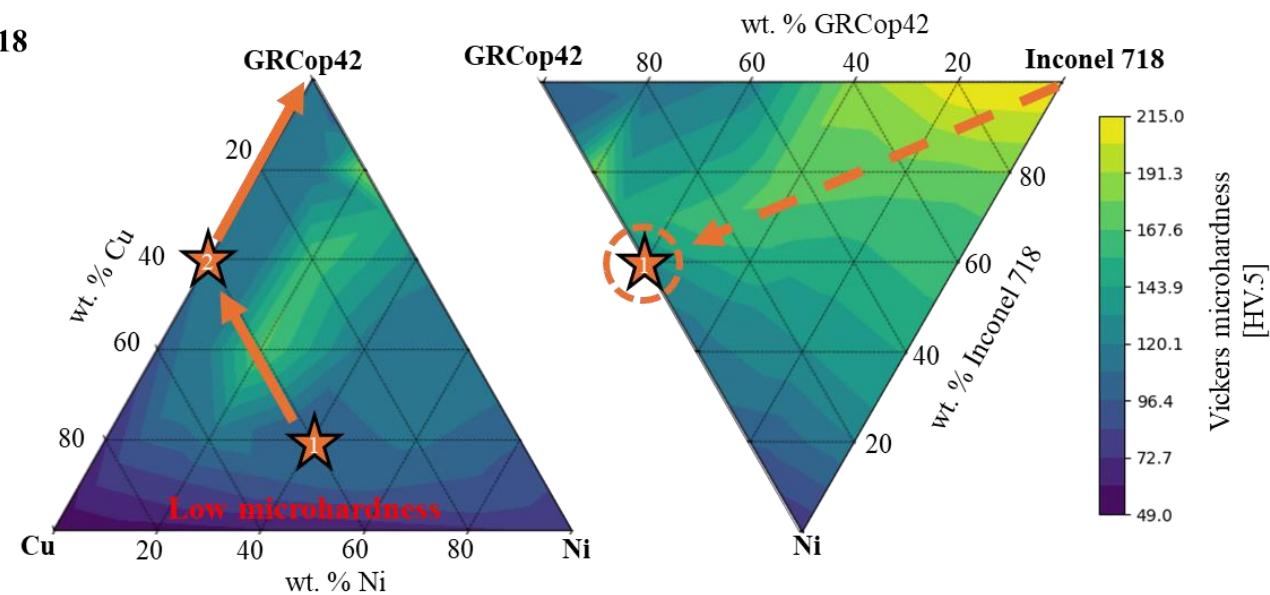
# Criterion for choosing transition compositions

1. Avoid regions with brittle phase formation.
2. Avoid regions of low strength.
3. Choose as few transition compositions as possible.

CALPHAD + Arc melted samples



Experimental microhardness results for mechanical property prediction



# Printing the joint

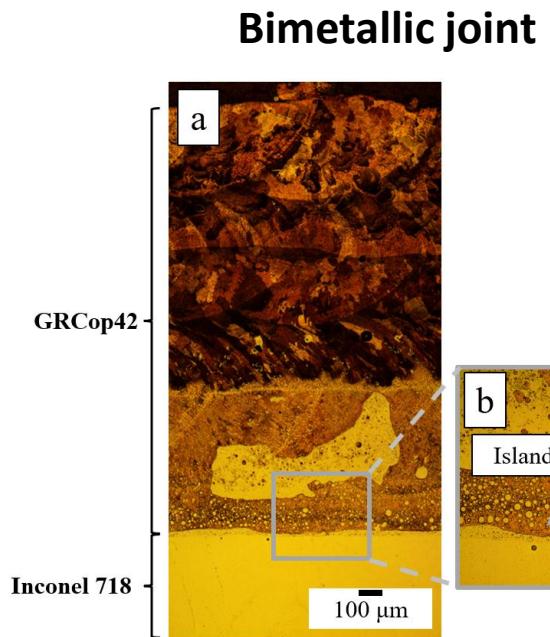
- Transition compositions created for LDED by mechanical mixing Cu, Ni, and GRCop42 powder.
  - Transition 1: 20 wt.% GRCop42, 40 wt.% Ni, 40 wt.% Ni
  - Transition 2: 60 wt.% GRCop42, 40 wt.% Cu
- Joint printed on Meltio M450
- Parameters adjusted in-situ for avoidance of defects.
- Speed of 400 mm/min and powder feed rate of 2 g/min was used for all powder printing.



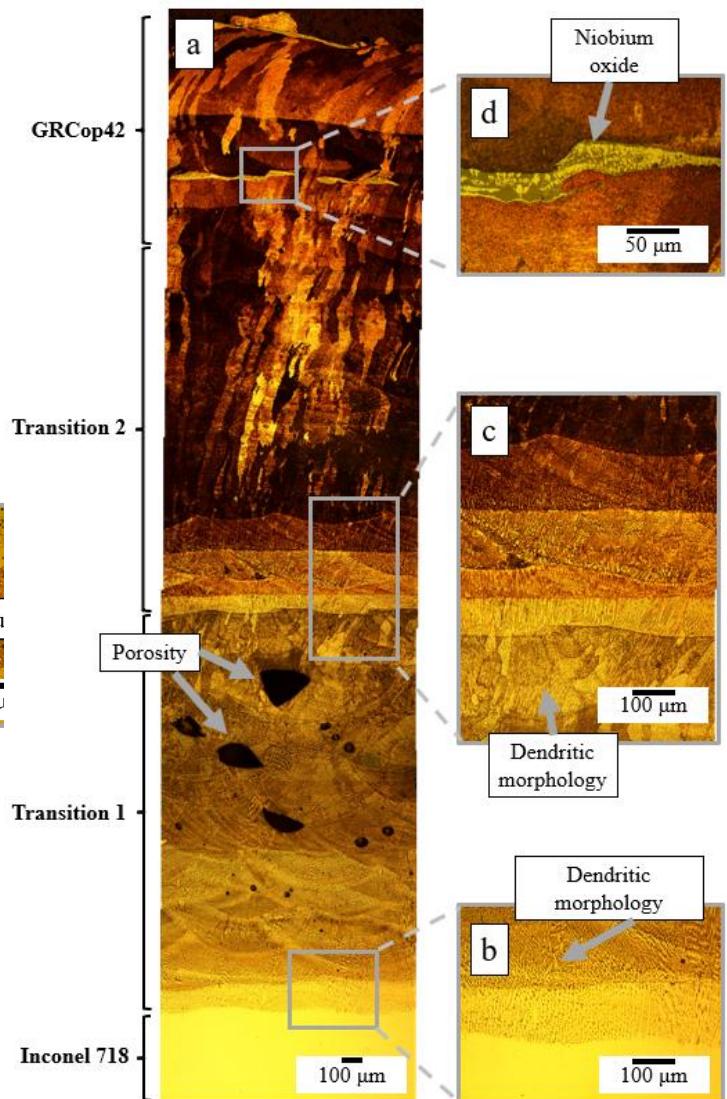
Material	Layer	Laser Power [W]
Inconel 718 wire	All	700
Transition 1	All	500
Transition 2	1-2	500
	3-5	600
	6	700
	7-8	800
	9	900
	10	1000
	1-3	1050
GRCop42	4-6	1100

# Optical micrographs

- Optical microscopy of joint shows no evidence of islands / peninsulas: **No brittle phases**
- Trapped gas porosity found in the filler 1 composition: attributed to powder mixing.
- **NbO** found in the GRCop42: attributed to high O<sub>2</sub> content.



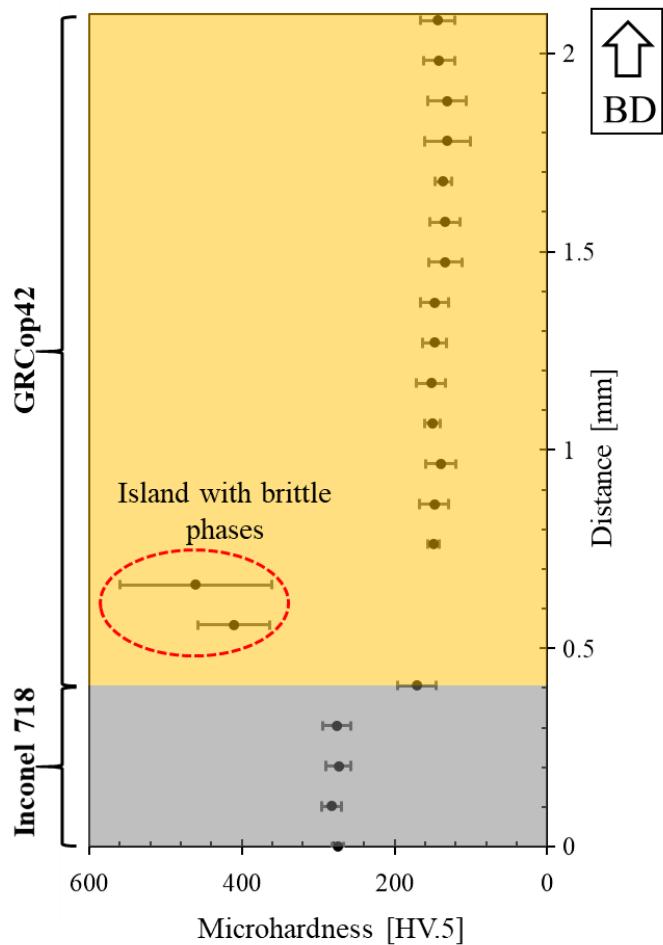
## Joint with transition compositions



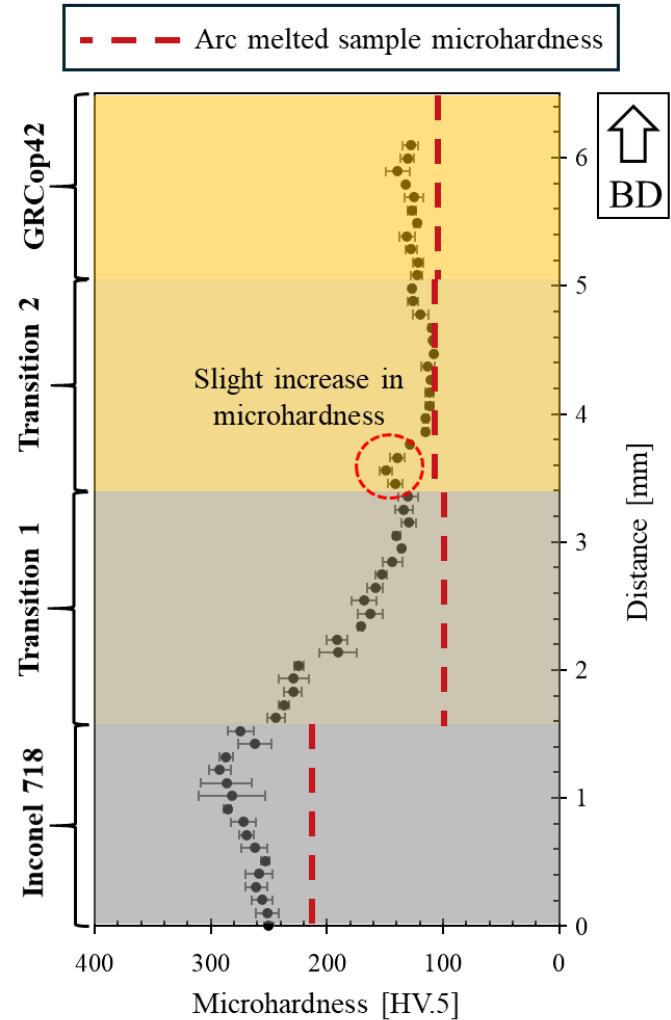
# Microhardness

- Microhardness values of compositions in the LDED joint found to be higher than arc melted counterpart.
  - Attributed to higher cooling rate / decrease in grain size of the LDED joint.
- Gradient of microhardness observed at each compositional change.
  - Attributed to remelting and mixing between the deposited layer and substrate layer.
- Lowest microhardness observed in Filler 2.
  - 12.7% lower than for GRCop-42.

Bimetallic joint



Joint with transition compositions



# Conclusions

- Investigated the effect of travel speed on wire – LDED Inconel 625.
  - Higher travel speed increases yield when pulled parallel to BD; does not increase yield when pulled perpendicular to BD.
- Investigated metallurgy of Inconel 625 – GRCop42 mixtures
  - Liquid miscibility gap leads to a Cu-deprived liquid solidifying to form brittle crack inducing phases.
- Investigated bimetallic joining of Inconel 625 – GRCop42
  - Inconel 625 can be deposited onto GRCop42 without brittle phases, but depositing GRCop42 onto Inconel 625 results in brittle phase formation.
- Developed transition compositions to join Inconel 718 with GRCop42
  - Transition compositions avoid brittle phase formation.

# Acknowledgements

- Advisor: Somayeh Pasebani
- Committee:
  - Dr. David McIntyre
  - Dr. Brian K. Paul
  - Dr. Julie Tucker
  - Dr. Donghua Xu
- Parents
- Dr. Peter Eschbach and Dr. Nick Wannenmacher
- Lab group and friends.



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# Questions?



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# Backup Slides

# Calculating thermal gradient and solidification rate

$$R = \frac{VD}{L}$$

$R$  – solidification rate

$V$  – travel speed

$D$  – depth of melt pool

$L$  – length of melt pool

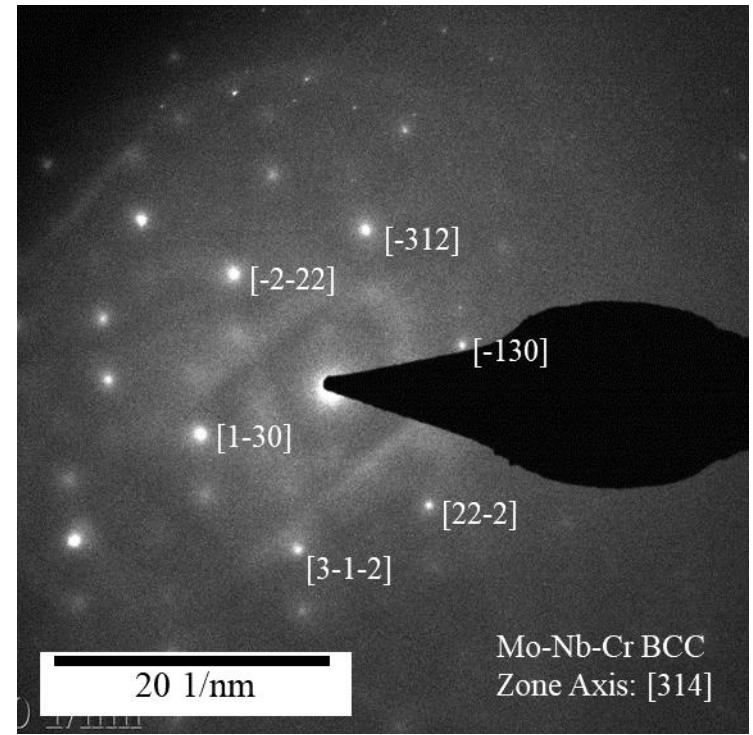
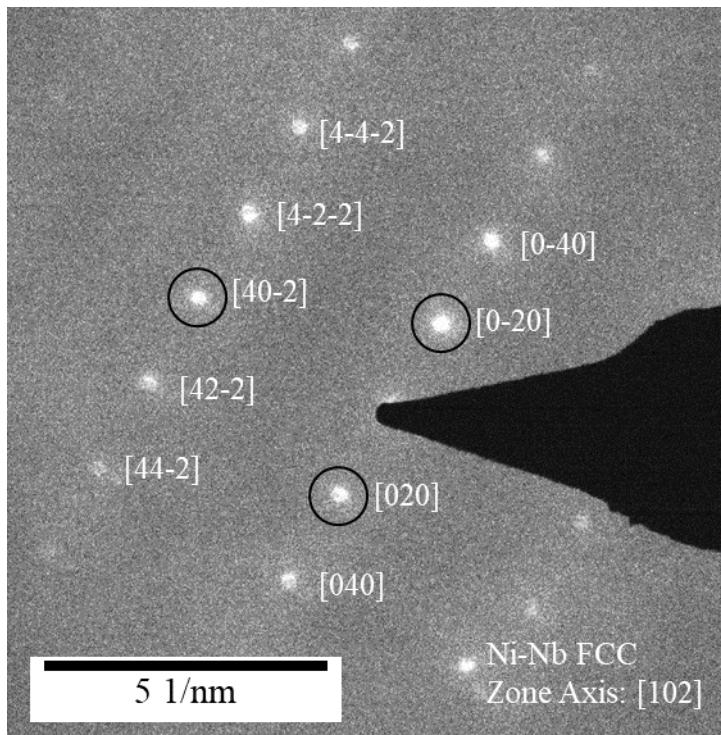
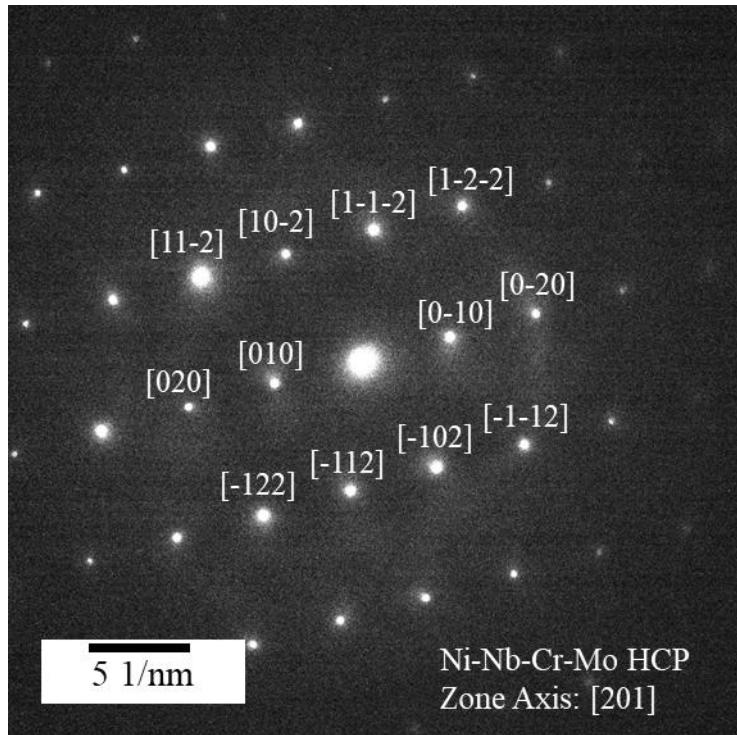
$$G = \frac{C_l}{R}$$

$G$  – thermal gradient

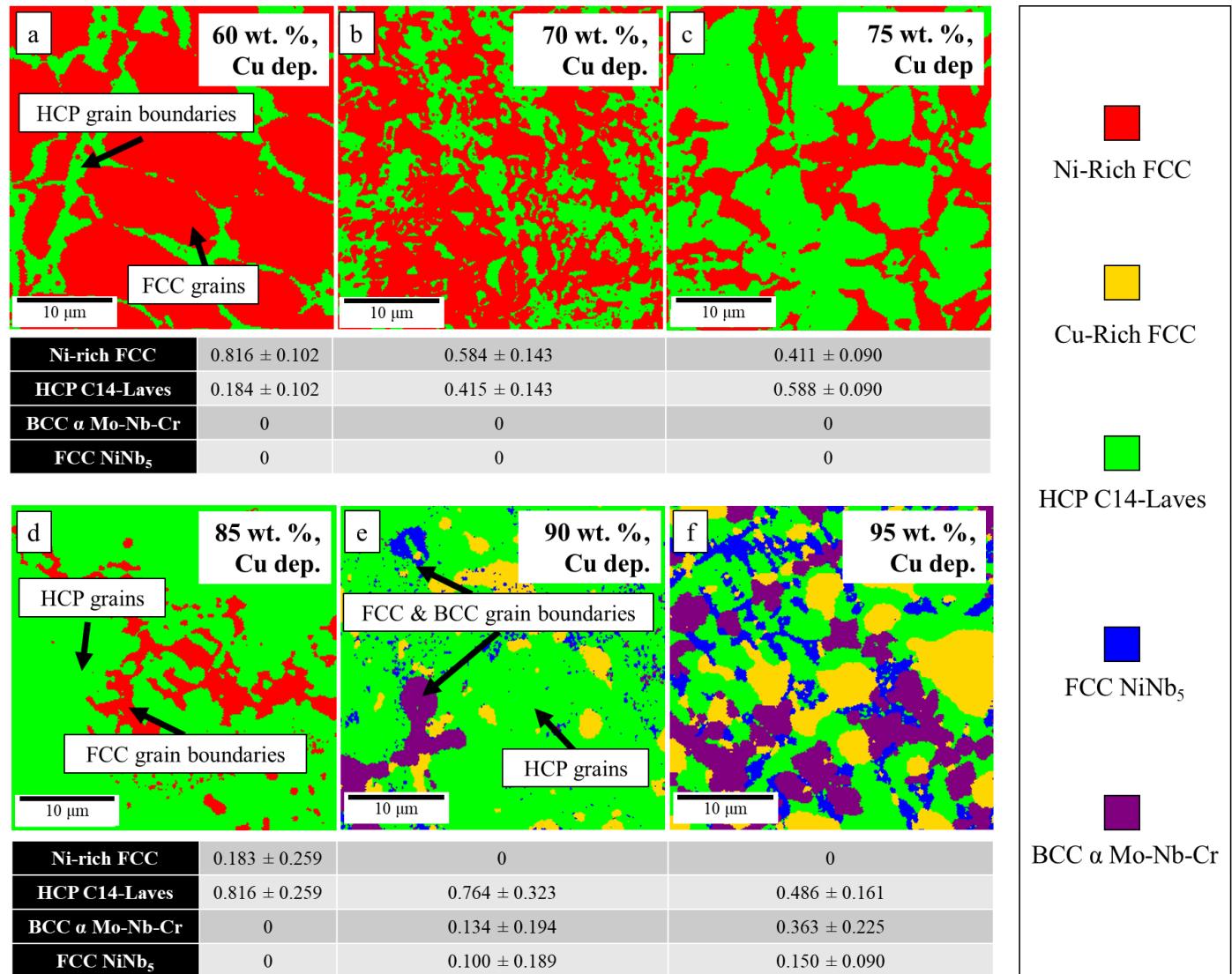
$C_l$  – cooling rate

$R$  – solidification rate

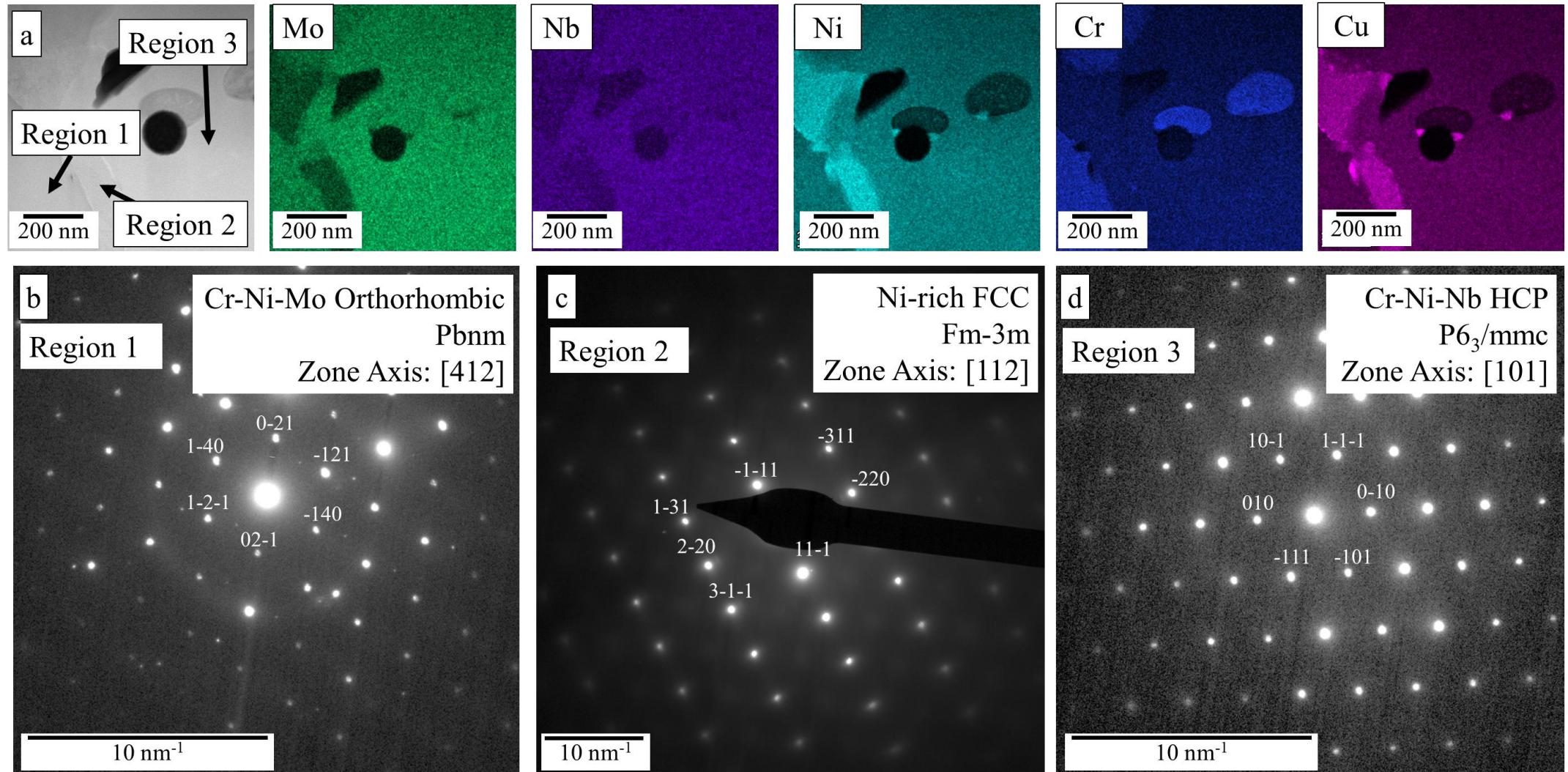
# SAD patterns of arc melted samples



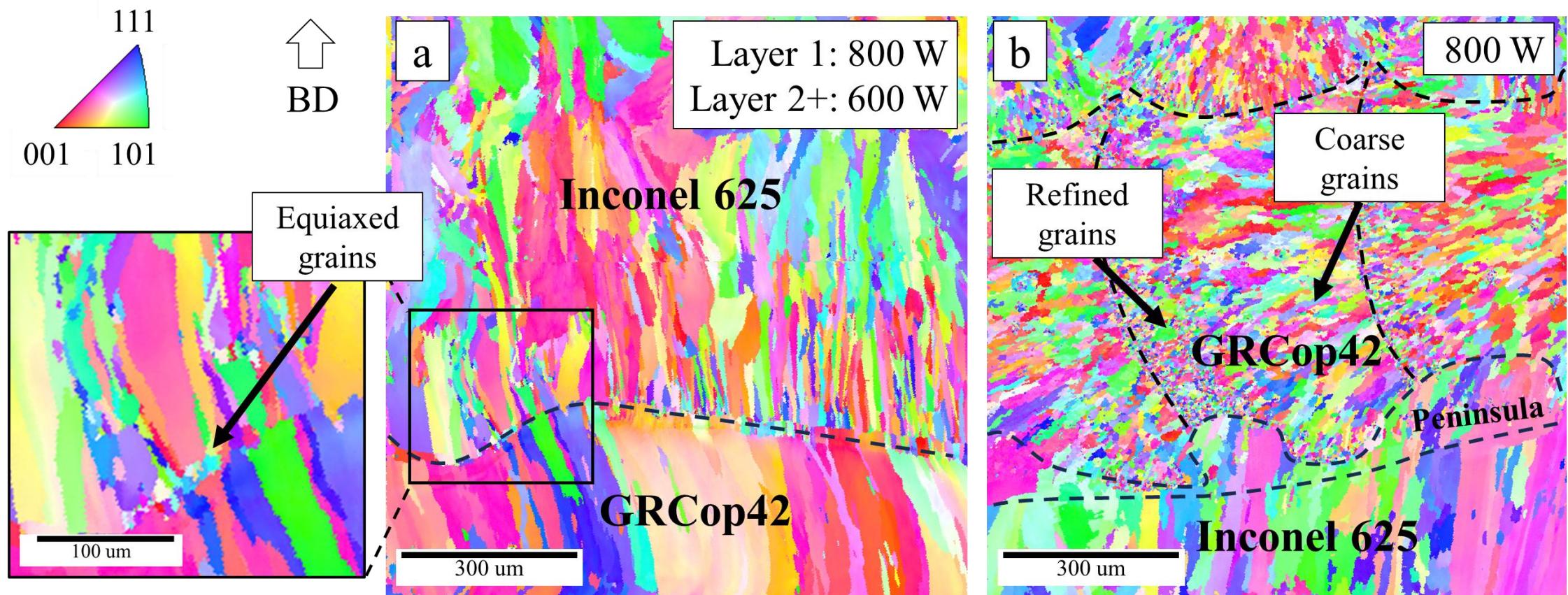
# EBSD phase maps of arc melted samples



# SAD patterns of LDED samples



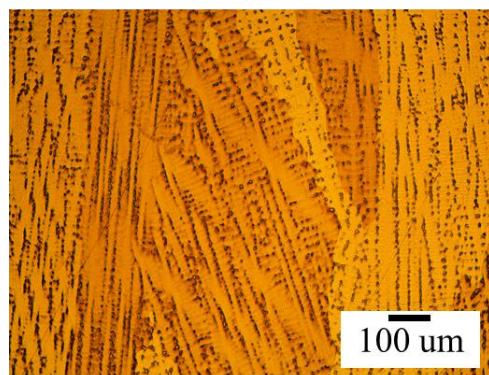
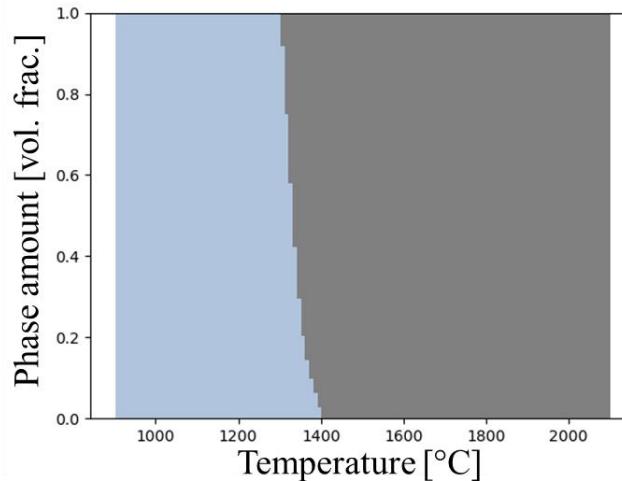
# EBSD of bimetallic joint



# Backup: Solidification paths

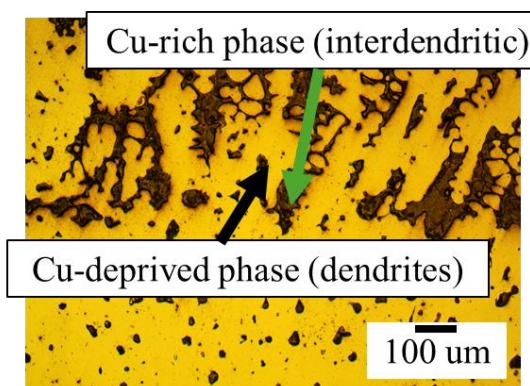
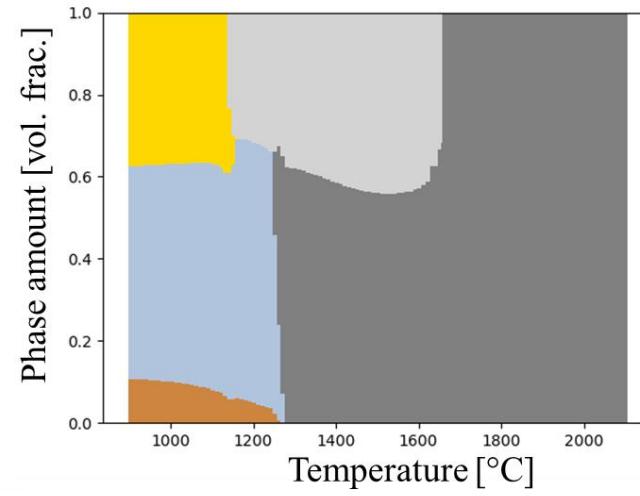
**Solidification path [A]:**

60 wt. % Ni  
20 wt.% GRCop42  
20 wt.% Inconel 718



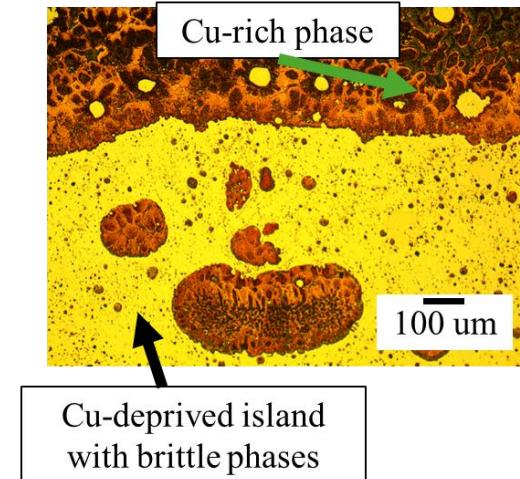
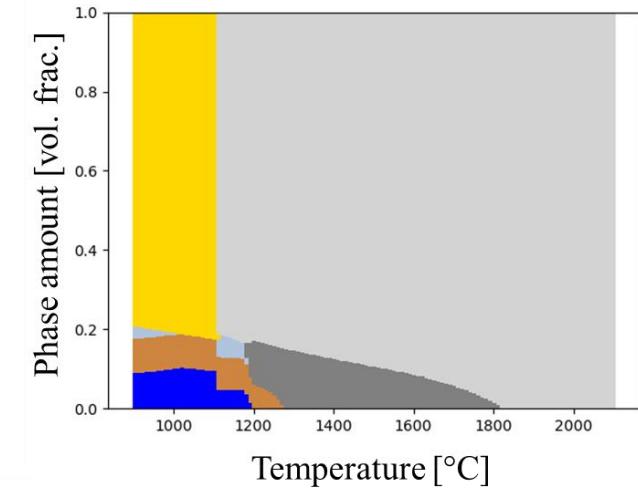
**Solidification path [B]:**

40 wt.% GRCop42  
60 wt.% Inconel 718



**Solidification path [C]:**

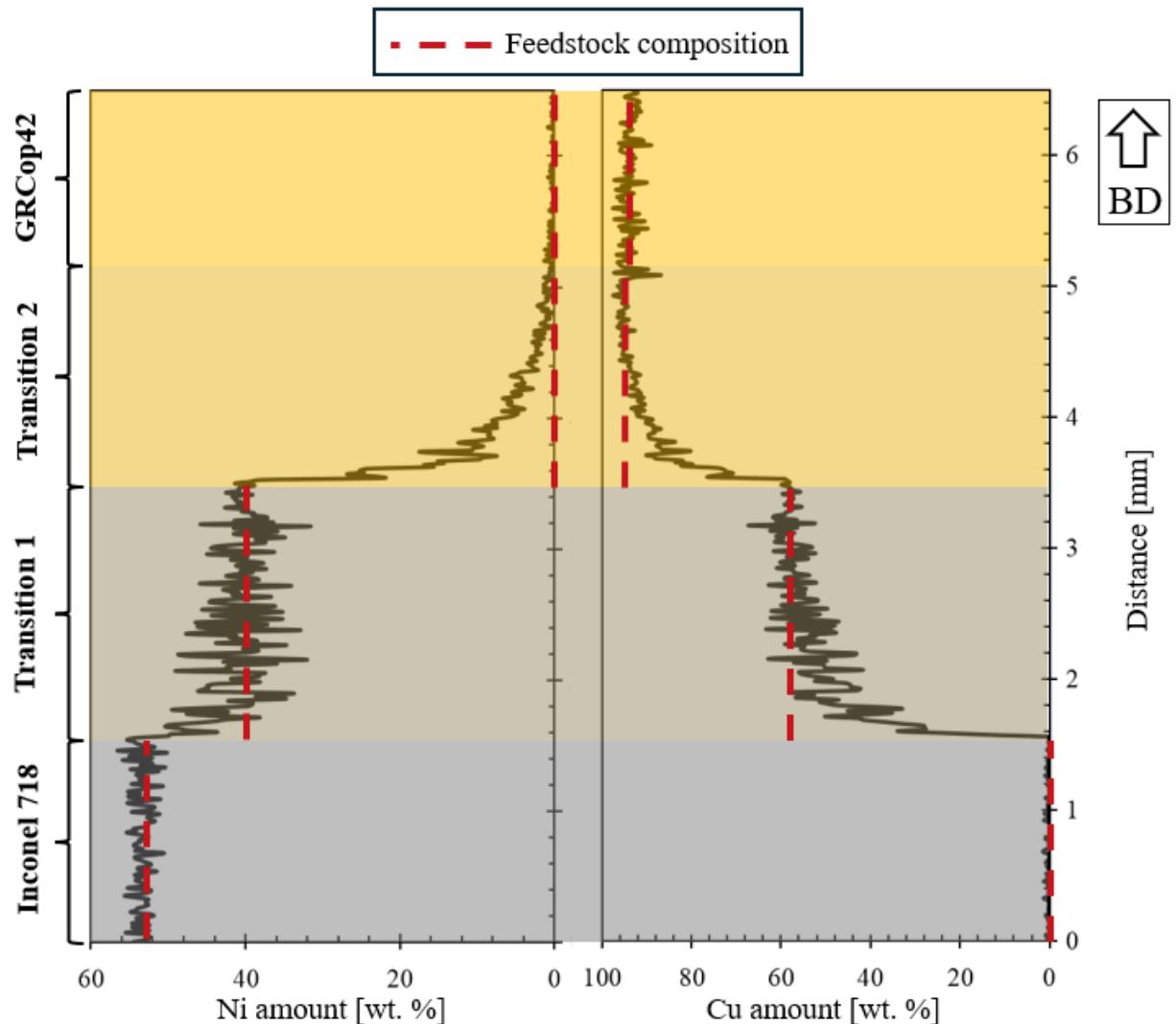
80 wt.% GRCop42  
20 wt.% Inconel 718



- █ Cu – rich FCC
- █ Cu – deprived FCC
- █ NbNi<sub>3</sub> ( $\delta$ )
- █ BCC\_B2
- █ Cu – rich liquid
- █ Cu – deprived liquid

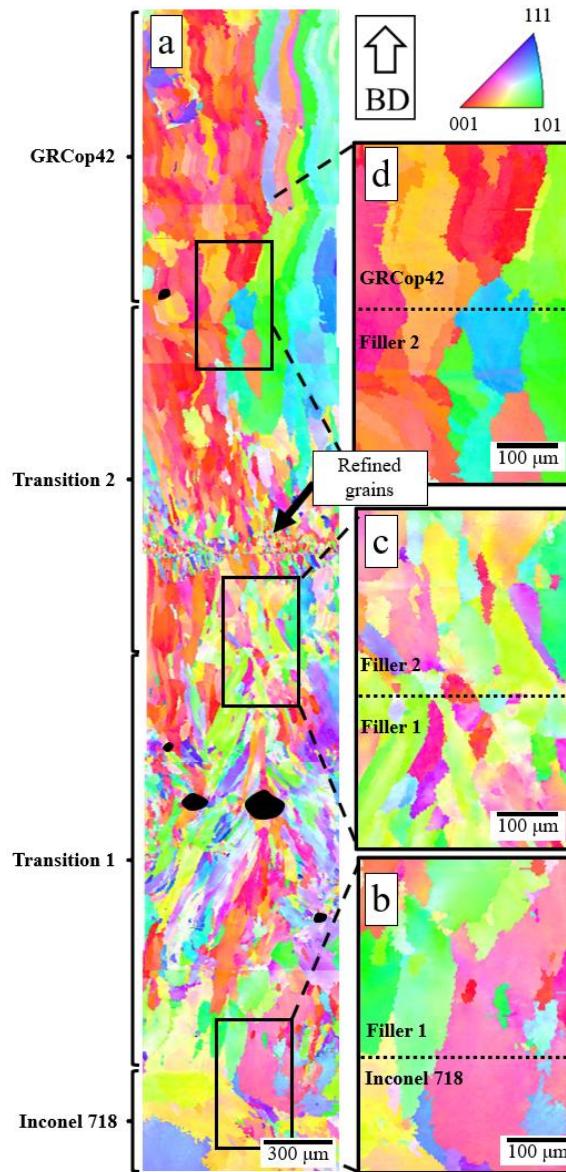
# Compositional profile

- Deposited feedstock matches target composition towards the top of each compositional region.
  - This is a prerequisite to avoid the liquid miscibility gap.
- Gradient observed at each compositional change.
  - Attributed to remelting and mixing between the deposited layer and substrate layer.



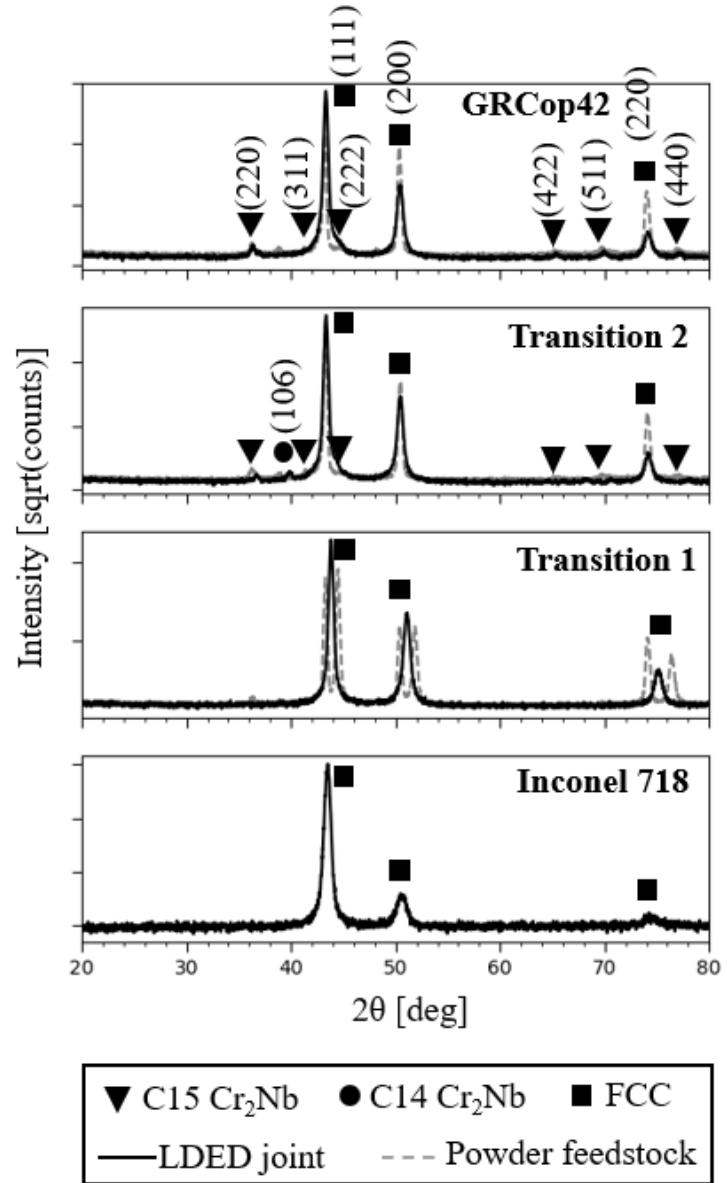
# Grain morphology

- Epitaxial growth observed at every compositional change.
  - This stands in contrast to Inconel GRCop-42 bimetal samples, which show a lack of epitaxial growth.
- Columnar grains observed throughout most of the joint except above the filler 1 – filler 2 transition.
  - Attributed to an increased cooling rate in this layer.
- Large columnar grains observed in GRCop-42.
  - Attributed to high laser power utilized.



# Present phases

- Only FCC phases detected in the Inconel 718 Transition 1 regions.
- Expected C15 Cr<sub>2</sub>Nb precipitates observed in the transition 2 GRCop42 regions.
- One peak corresponding to C14 Cr<sub>2</sub>Nb observed.
  - This could be attributed to Ni migration at the interface.



# Inconel 625 onto GRCop42



- When depositing Inconel 625 onto GRCop42 at 600 W, lack of fusion observed in the first layer due to the GRCop42 acting as a heat sink.
- The lack of fusion is avoided by increasing the laser power in the first layer.
- Defect-free joint achieved – miscibility gap avoided by remelting less than 20 wt.% GRCop42.

