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# Plasmons in double interfaces system of $Si_3N_4/SiO_2/Si$ irradiated by $^{60}Co$

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#### Abstract

The first level plasmons of Si in the pure Si state, in the  $SiO_2$  state and in the  $Si_3N_4$  state (corresponding to bonding energy 116.95, 122.0 and 127.0 eV) were investigated directly with X-ray photoelectron spectroscopy before and after  $^{60}$ Co radiation. The experimental results demonstrate that there existed two interfaces, one consisted of plasmons of Si in the  $Si_3N_4$  and  $SiO_2$  states, while another was made of plasmons of Si in the pure Si state and in the  $SiO_2$  state. When the  $Si_3N_4$ – $SiO_2$ –Si samples were irradiated by  $^{60}$ Co, the interface at  $Si_3N_4$ / $SiO_2$  was extended and at the same time the center of this interface moved towards the surface of  $Si_3N_4$ . The concentration of plasmon for silicon in the  $SiO_2$ –Si interface are observable. Finally, the mechanism of experimental results is analyzed by the quantum effect of plasmon excited by the photoelectron.

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#### 1. Introduction

The SiO<sub>2</sub>/Si interface has been extensively studied in the past years because of its influence on the performance of metal—oxide—semiconductor devices, especially for metal—oxide—semiconductor-very large-scale integrated (VLSI) circuits. The wear out and breakdown of thick SiO<sub>2</sub> film has been extensively researched. Several models involving local hole trapping and oxide field buildup during Fowler—Nordheim electron tunneling have been proposed to explain oxide breakdown [1,2]. A lower effective barrier height for electrons and the image charge effect are caused by

positive oxide charge (hole trapping) and tunneling path of Fowler–Nordheim reduced [2,3]. Because of lack of a physics-based model, there has been persistent concern over the severity and even the existence of positive oxide charge buildup, since the power supply voltage decreased from 5 to 3.4 V.

Several mechanisms were presented to explain charge generation by ionic radiation [3–6]. For thick oxides the impact ionization model of electron–hole pair generated in the oxide bulk shows reasonable agreement with experiment. However, the generation mechanism of the radiation damage induced by  $^{60}$ Co in double interfaces system of Si<sub>3</sub>N<sub>4</sub>/SiO<sub>2</sub>/Si has not yet been studied, especially, using some ways to detect plasmon. Attempts to detect the surface and bulk plasmons

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can be on indirect observations, such electronenergy loss, radioactive decay of the plasmons, or measurement of  $\mathrm{d}I^2/\mathrm{d}V^2$  [7]. It is X-ray photoelectron spectroscopy (XPS) that can give signal of the surface, interface and bulk plasmons directly. In this study, double interfaces system of  $\mathrm{Si}_3\mathrm{N}_4/\mathrm{SiO}_2/\mathrm{Si}$  sample irradiated by  $^{60}\mathrm{Co}$  was measured by the XPS; focus is on the changes of the interfacial plasmons created by radiation condition. The experimental results demonstrate that there existed such variance, and the results are explained by the quantum effect of plasmons excited by photoelectron.

#### 2. Experimental details

The structure of the Si<sub>3</sub>N<sub>4</sub>/SiO<sub>2</sub>/Si is shown in Fig. 1. The samples utilized in this work were in the form of silicon wafer (n-type  $\rho$  6–9  $\Omega$ /cm) with surface orientation  $\langle 100 \rangle$ . Oxide was grown in dry O<sub>2</sub> at 1273 K to a thickness of 21 nm firstly, then 40 nm silicon nitride layer was deposited on top of the oxide by a low pressure CVD and annealed in situ in at mix gas of oxygen and hydrogen  $(O_2 + H_2)$  at 1173 K for 30 min resulting in Si<sub>3</sub>N<sub>4</sub>/SiO<sub>2</sub>/Si sample. Because the thermal expansion coefficients of SiO2, Si and Si3N4 are different, annealing at high temperatures may create some interfacial defects. In order to keep the level of interfacial defects was the same for all samples before radiation, all samples were taken from one Si<sub>3</sub>N<sub>4</sub>/SiO<sub>2</sub>/Si chip and were p-channel transistors with metal Al gates. In such case, however, the experimental results would not be affected to follow comparing of among the spectra of plasmon in the form of XPS. The thinner oxide is, the lower sensitive to ionization radiation of

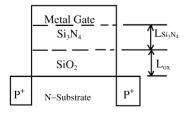


Fig. 1. The structure of the  $Si_3N_4$ – $SiO_2$ –Si measured in this work.

 $SiO_2/S$  interface will be [8,9]. According to this reason, here the oxide was grown in thin form.

The  $\gamma$ -irradiation was performed at room temperature with a  $^{60}\text{Co}$  source of an energy of 1.25 MeV, and the dosage of radiation ranged from 0 to  $10^4$  Gy (Si). The bias fields during irradiation were 0 and -1 MV/cm. In order to make targets appropriate for XPS measurement, hydrogen peroxide and sulphuric acid solution was employed to remove the Al layer of Ohmic contact before inserting the Si $_3$ N $_4$ /SiO $_2$ /Si samples into the XPS cavity.

XPS measurements were carried out via a KRATOS XSAM800 electron spectrometer. An Ar + ion beam was used to etch the Si<sub>3</sub>N<sub>4</sub> and SiO<sub>2</sub> films. There might be two effects of sputtering. Firstly, when sputtering, there was always the possibility of different sputtering rates of the elements, another is that atoms might be pushed down into deeper layers of the sample and atoms from the lower layers of the sample might be transported to the upper layers of the sample. In order to overcome the effects above on experimental results, each sample was etched by a fixed ion beam parameters of Ar<sup>+</sup>, i.e., at energy of 4 keV, identical etching time and beam current to keep to the same extent of the two effects upon each sample. In such case, however, the experimental results would not be affected to follow comparing of among the XPS spectra. The vacuum during the measurement was less than  $1.3 \times 10^{-6}$  Pa. The samples were excited by 1486.6 eV Al K<sub>a</sub> X-rays. The sweep range was 20 eV for both Si<sub>3</sub>N<sub>4</sub> and SiO<sub>2</sub>, and other parameters of measurement, such as reg, step, dwell and scan numbers were the same for every measurement. Each spectrum was calibrated to C at bonding energy (BE) 285 eV, and every peak was stimulated from original spectrum by code DS300X Data System. BEs have been used as index of the plasmons [10,11].

### 3. Results and discussion

In order to voice interfaces objectively, relative intensity is accepted. Fig. 2 gives plots of the etch time-normalized area before and after radiation

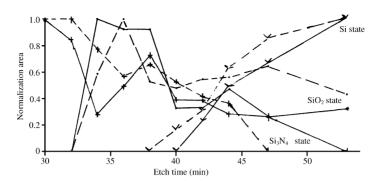


Fig. 2. Profiles of the etch time—normalized area for plasmon in the  $Si_3N_4/SiO_2/Si$  before and after radiation. The curves - - - and —refer to results of plasmon from the  $Si_3N_4/SiO_2/Si$  samples unexposed and exposed to  $10^4$  Gy (Si) of  $^{60}$ Co gamma radiation at -1 MV/cm bias field, respectively. The direction of normalized areas for plasmons in the pure Si state ( $^{\times}$  BE 116.95 eV), in the  $Si_3N_4$  state (+ BE 127 eV) and in the  $SiO_2$  state( $\bullet$  BE 122 eV) are area at the Si substrate, area at the surface of  $Si_3N_4$  and the peak with maximal area of plasmon in the  $SiO_2$  state.

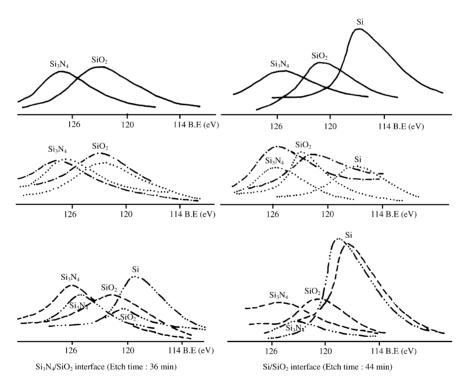


Fig. 3. The plasmon peaks fitted from original spectra at the  $Si_3N_4/SiO_2$  and  $SiO_2/Si$  interfaces irradiated in different radiant environment. The curves expressed by —, • • • •, —•—• and — — refer to samples exposed to  $0.10^2.10^3$  and  $10^4$  Gy (Si) of  $^{60}$ Co gamma radiation at -1 MV/cm bias field, — • • • — • • • is the sample exposed to  $10^4$  Gy (Si) of  $^{60}$ Co gamma radiation at 0 MV/cm bias field (before being referenced to C).

[10]. The direction of normalized areas for plasmons in the pure Si state (BE 116.95 eV), in the  $Si_3N_4$  state (BE 127 eV) and in the  $SiO_2$  state

(BE 122 eV) are area at the Si substrate, area at the surface of  $Si_3N_4$  and the peak with maximal area of plasmon in the  $SiO_2$  state. Because the sputter

time corresponds to the depth from the surface of Si<sub>3</sub>N<sub>4</sub> to the substrate Si, Fig. 2 also shows the profiles of plasmons in the Si<sub>3</sub>N<sub>4</sub>/SiO<sub>2</sub>/Si. It can be observed in Fig. 2 as expected that (1) there exists an interface of plasmons made of Si<sub>3</sub>N<sub>4</sub> and SiO<sub>2</sub>, and another interfaces of plasmons consisted of SiO<sub>2</sub> and Si. Furthermore, if the sputtering effect level of Ar<sup>+</sup> ion beam on Si<sub>3</sub>N<sub>4</sub> is considered to be similarity to that on SiO2, the thickness of the interface Si<sub>3</sub>N<sub>4</sub>-SiO<sub>2</sub> is about the same that of SiO<sub>2</sub>-Si. (2) The interface of plasmons at Si<sub>3</sub>N<sub>4</sub>/ SiO<sub>2</sub> is extended, meanwhile, the center of this interface moved towards the surface of Si<sub>3</sub>N<sub>4</sub> by the actions of radiation. Unlike previous work [12], the interface of plasmons at SiO<sub>2</sub>/Si is not changed.

Fig. 3 displays the differences in plasmons of the double interface samples in the XPS spectra before and after radiation. The figure shows that the feature of plasmon labeled "Si" in the  $SiO_2$  state located at the interface  $SiO_2$ –Si was decreased when the radiation dose was up to  $10^4$  GY (Si) at the bias field of -1 MV/cm, obviously, and the plasmon of Si in the  $SiO_2$  state reduced greatly due to the bias-free field at this radiation dose.

To explain these results, the photoelectron energy loss in Si<sub>3</sub>N<sub>4</sub> and SiO<sub>2</sub> is discussed by using Fig. 4. When a beam of photons is brought to bear on a target, energy is deposited in the materials as kinetic energy and the potential energy of the atoms ionized and excited. The energy of photoelectron generated by <sup>60</sup>Co is above 80 eV at least. The photoelectron of this kind can slightly penetrate into the Si<sub>3</sub>N<sub>4</sub> and SiO<sub>2</sub> because the inelastic mean free path (IMFP) of photoelectron is around 0.5 nm. Then, photoelectron excite surface and interfacial plasmons as well as secondary electrons, which can excite bulk plasmon by the "spur" process [13]. This energyloss process is called the "blob" process. The characteristic length of the reaction volume in which plasmons are excited by electron is known as radius  $d_r$ . The  $d_r$  for electron above 80 eV is above 10 nm, and called the Onsager radius.

It is known that only the photoelectrons with energy equal to  $nhv_s$ ,  $mhv_i$  or  $khv_b$  can lose their energy by exciting surface plasmons, interfacial plasmons and bulk plasmons, respectively. Here,

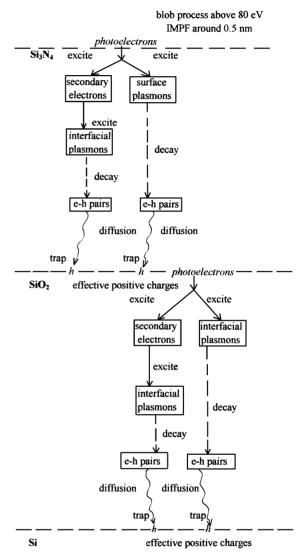


Fig. 4. Photoelectron energy-loss theory in  $Si_3N_4/SiO_2$ . Blob process: Photoelectrons above  $80\,\text{eV}$  can slightly penetrate into the  $Si_3N_4$  and  $SiO_2$  because the IMFP is around 0.5 nm. Then, photoelectrons excite surface plasmons, interfacial plasmons as well as secondary electrons, which can create interfacial plasmons by the spur process. The  $d_r$  for electrons above  $80\,\text{eV}$  is about  $10\,\text{nm}$ .

n, m and k are integers, and  $hv_s$ ,  $hv_i$  and  $hv_b$  are the quantum creation energies of surface plasmon, interfacial plasmon and bulk plasmon, respectively. The quantum absorption energy can be calculated theoretically [14,15]. The path to create plasmon of Si in the SiO<sub>2</sub> state is the shortest in

samples, and the yield of photo-induced electrons is  $10^{-4}$  order. Furthermore, the direction of acceleration for photoelectron due to  $-1 \,\mathrm{MV/cm}$ bias field is from the surface of Si<sub>3</sub>N<sub>4</sub> to Si substrate, but the oxygen atoms either couple with N or do Si in SiO<sub>2</sub> layer. The intensity fraction of plasmon for Si in SiO<sub>2</sub> state may be very small generated by photoelectrons and secondary electrons due to reasons above, parallely, therefore, the intensity percentage of plasmon for Si in SiO<sub>2</sub> state had not been changed until the dose was up to 10<sup>4</sup> GY (Si). Most photoelectrons create plasmons either in the layer of Si<sub>3</sub>N<sub>4</sub> or in the layer of Si and are without loss energy in the layer of SiO<sub>2</sub>. If the dose is up to  $10^4$  GY (Si) with higher dose rate and acceleration field, which looks like pumping effect, thus, there is the smallest intensity fraction of plasmon corresponding to SiO<sub>2</sub> in the XPS spectra. The feature labeled plasmons produced by 60Co at bias-free field is substantially more intense of Si in the pure Si state. The possible reason is that the energy of photoelectron excited by gamma ray at bias-free field may be the lowest in this experiment, as well as the quantum creation energy of bulk plasmon  $hv_b$ , and these make choice absorption easy. When one pay attention to the change of profile, the photoelectrons transmitted Si<sub>3</sub>N<sub>4</sub> and the layer of this medium is longer than that of SiO<sub>2</sub>, this is one reason why the interface of plasmons at Si<sub>3</sub>N<sub>4</sub>/SiO<sub>2</sub> is extended and the center of this interface moved towards the surface of Si<sub>3</sub>N<sub>4</sub>, or this result suggests that the existion of the Si<sub>3</sub>N<sub>4</sub> layer replaced the main area of plasmons generated from the SiO<sub>2</sub> layer into Si<sub>3</sub>N<sub>4</sub> layer.

#### 4. Conclusions

The plasmons of the double interfaces system of  $Si_3N_4/SiO_2/Si$  have been analyzed before and after

radiation, successfully. The plasmons are excited in zone of  $Si_3N_4$  and Si mainly, this is very unlike the plasmons produced by gamma rays in the interface of  $SiO_2/Si$ . The bias field is still very important factor among radiation environment. Experimental results also demonstrated that in order to reduce the degeneration of MOS generated by ionic radiation, the introduction of  $Si_3N_4$  layer is necessary because it transferred the main area to crate plasmons from  $SiO_2/Si$  interface into  $Si_3N_4$ – $SiO_2$  interface.

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