

1 **By how much can closed loop frameworks**
2 **accelerate computational materials discovery?**

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8 (Dated: September 30, 2022)

Abstract

Implementation of automation and machine learning surrogatization into closed-loop computational scientific workflows is an increasingly popular approach to accelerate materials discovery. However, the scale of the associated speedup from this paradigm shift over traditional manual approaches remains an open question. In this work we rigorously quantify the acceleration from each of the components within a closed-loop framework by probing four sources of speedup: (1) automation, (2) calculation runtime improvement, (3) guided design space search, and (4) machine-learning surrogatization. This is done through the timing of automated software and corresponding manual computational experiments. Stemming from a combination of the first three speedup sources, we estimate that acceleration of materials discovery by over $10\times$ can be achieved. By introducing surrogatization into the loop, we estimate that this can be further improved to $15\text{--}20\times$. This work highlights the value in closed-loop approaches towards accelerating materials discovery.

9 Keywords: automated high-throughput DFT, sequential learning, computational materials discovery

10 I. INTRODUCTION

11 Discovery of materials is a central barrier to next-generation energy technologies such
12 as more efficient and environmentally friendly electrochemical synthesis processes. One
13 particular example is to substitute the environmentally harsh Haber-Bosch process used to
14 synthesize ammonia by identification of candidate materials that can catalyze the reaction
15 electrochemically [1, 2]. However, finding such optimal candidates efficiently remains a
16 challenge due to the large size of the feasible candidate space [3]. Development of methods
17 to accelerate this search, even within a relatively bounded design-space, is crucial to meet
18 approaching climate goals.

19 These considerations have motivated significant research into new methods for accel-
20 ated materials discovery, both experimentally and computationally [4, 5]. In the context of
21 experimental screening, much research focus has taken the form of robotic experimentation
22 for applications such as searching for battery electrolytes [6], finding thermally stable per-
23 ovskites [7], and optimizing battery charging protocols [8]. These studies tend to employ a

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24 combination of robots to automate experimental tasks and a learning agent to guide sub-
25 sequent studies, closing the loop. However, the trade-off is that automated experimental
26 setups are highly application specific and difficult to adapt to new applications where dif-
27 ferent tasks may be required. Thus experimental workflows show much promise, but are at
28 present limited in terms of generalizability.

29 In contrast, fully computational workflows are appealing due to being mainly compute-
30 limited and are relatively more flexible in terms of application. These workflows share some
31 similarities to closed-loop experimental workflows, with computational calculations substi-
32 tuted for experiments and algorithms for iteratively selecting candidates from the design
33 space. Adding new tasks to computational workflows demands only additional compute
34 resources, rather than physical materials necessary to synthesize and test new candidates.
35 This allows for improved modularity for transferring existing closed-loop software compo-
36 nents between different materials workflows. The use of an iterative guided design space
37 search has demonstrated encouraging results in terms of speeding up materials discovery
38 [9–25]. Consequently, multiple computational closed-loop workflows have been developed
39 for applications such as catalyzing electrochemical CO₂ reduction and hydrogen evolution
40 [26], finding stable iridium oxide polymorphs [27], and discovering stable binary and ternary
41 systems [28]. In each of these studies, the guided candidate searches showed notable im-
42 provement in finding promising candidate materials over a random search.

43 While computational closed-loop frameworks demonstrate a promising approach to accel-
44 erate materials discovery, quantification of their benefits over more traditional approaches
45 remains challenging. In particular, the degree to which cumulative speedups of a fully au-
46 tonomous closed-loop framework combine to accelerate materials discovery remains unclear.
47 To our knowledge, a detailed breakdown of sources of acceleration along with relative quan-
48 titative impacts on speedup has not been previously explored.

49 In this study we quantify the acceleration estimates of a closed-loop computational frame-
50 work for an electrocatalysis application. We probe two types of fully autonomous computa-
51 tional workflows (Figure 1): i) a closed-loop framework consisting of high-throughput density
52 functional theory (DFT) calculations which feeds into a sequential learning (SL) algorithm
53 that can select the next batch of candidate systems (thereby closing the loop) and ii) an
54 extension of the previous framework where the cycle has produced enough DFT data that a
55 machine learning (ML) surrogate can be trained to a desired accuracy and replace the heavy

56 DFT calculations. Four categories of acceleration are considered:

- 57 1. Comprehensive end-to-end automation of computational workflows
- 58 2. Runtime improvements of atomic compute tasks
- 59 3. Efficient search over vast design spaces using uncertainty-informed SL, and
- 60 4. Surrogatization of the most time-consuming tasks with ML models

61 Within each of these categories we estimate respective speedups and accumulate them into
62 overall acceleration metrics. For end-to-end streamlining we estimate the attributed speedup
63 through timing comparisons of the automated tasks and their manual analogues. In addition,
64 we introduce a human-lag model to simulate user-related delays associated with manual job
65 management on a computing cluster. Acceleration from improved runtimes of the compute-
66 tasks are presented in terms of both calculator settings and initial structure guesses for
67 DFT structural relaxations. This is done via calculations for relaxing OH onto the hollow
68 sites of a sample single-atom alloy, Ni₁/Cu(111). Efficiency improvements in design-space
69 searching are presented here via a simulated SL search on a sample problem of optimizing
70 binding energies of CO. Speedup from leveraging ML-surrogates for energy prediction is
71 estimated via DFT training set size needed to reach the desired model accuracy. Finally, we
72 then accumulate these results into an overall acceleration for workflows both excluding and
73 including surrogatization. Through a combination of improvements in each of the above
74 areas, we demonstrate a reduction in time to discover a new promising electrocatalytic
75 material by 80-95% when compared to conventional approaches.

76 II. RESULTS

77 Each of the forms of acceleration described above can synergize to provide overall speedup
78 in materials discovery. We benchmark the acceleration of each individual category through
79 timing estimates of the relevant components both within a closed-loop automated work-
80 flow and for equivalent tasks when using the traditional approach. For quantifying the
81 baseline materials discovery estimates, we use a combination of both modeling and man-
82 ual timing experiments. For the automated tasks, we use the *AutoCat* ([https://github.](https://github.com/aced-differentiate/auto_cat)
83 [com/aced-differentiate/auto_cat](https://github.com/aced-differentiate/auto_cat)) and *DFT-in-the-Cloud* (DFTitC) software packages

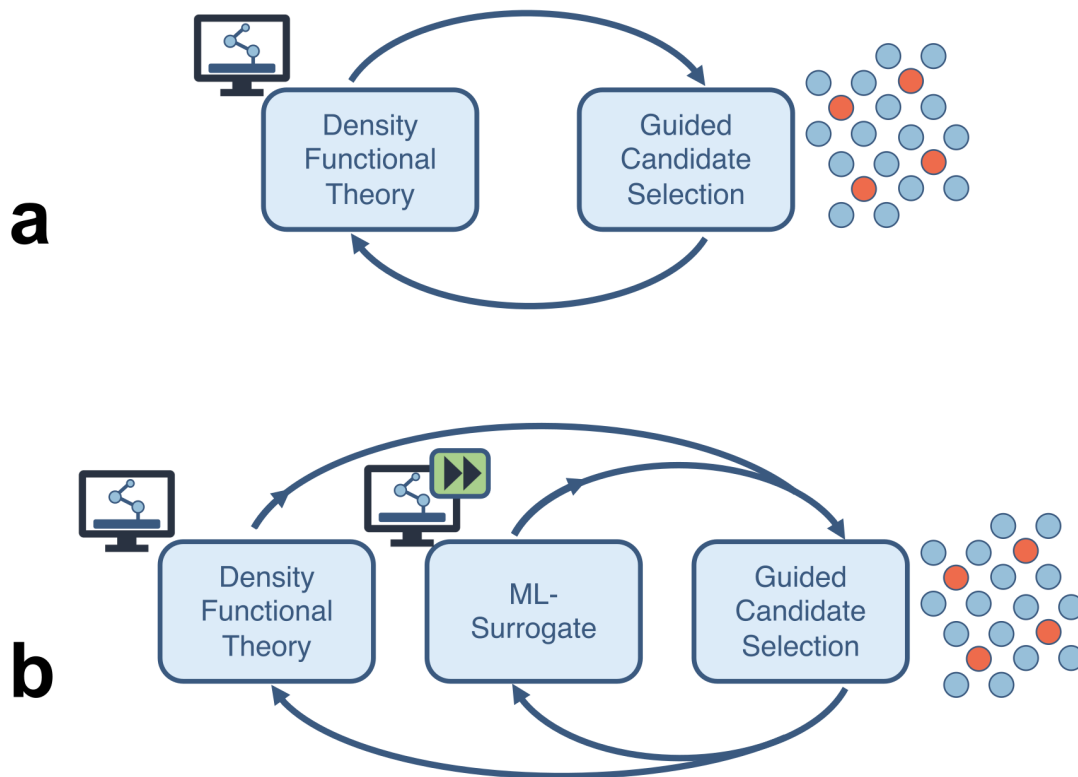


FIG. 1. Closed-loop materials discovery frameworks, a) without and b) with machine learning surrogates for the density functional theory calculations, to be considered in this work for acceleration quantification.

84 in tandem. Manual timing experiments use the *Atomic Simulation Environment* (ASE) [29]
 85 software package. Additional details are provided in Section V.

86 As an example design space, we chose the single-atom alloy (SAA) class of materi-
 87 als. SAAs are host transition-metals whose surface contains dispersed atoms of a different
 88 transition-metal species, and have shown much promise for electrocatalysis applications [30].
 89 In particular, we focused our efforts on probing SAA systems that can catalyze electrochem-
 90 ical ammonia synthesis.

91 In the following subsections we discuss each of the individual acceleration categories and
 92 how their estimates were obtained. This is followed by acceleration estimates of the full
 93 workflow combining all sources of speedup to obtain a single acceleration estimate from the
 94 automated approach relative to the traditional baseline.

A. Automation of Computational Tasks and Workflows

Within a standard computational study, there are many time-consuming tasks related to preparing, managing, and analyzing DFT calculations. In Figure 2 we visualize a typical pipeline for a computational electrocatalysis study. Each of the boxes underneath a head symbol represents a task where user involvement is required in the traditional paradigm. This includes structure generation, DFT pre- and post-processing, and job management. Thus, every box in the pipeline that relies on user intervention is an opportunity for streamlining through automation.

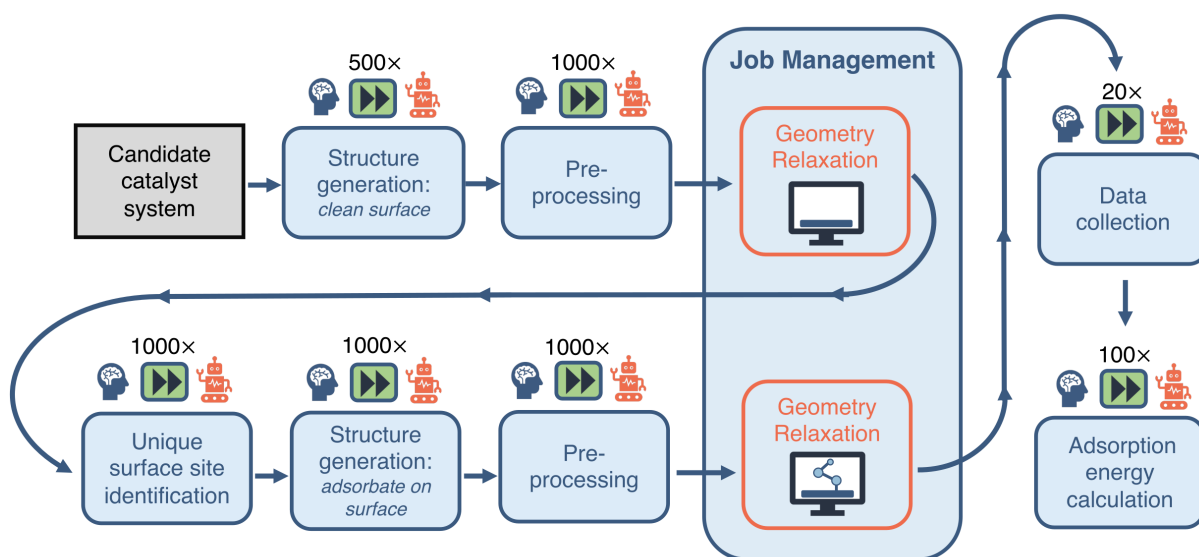


FIG. 2. Workflow for computational investigation of materials for electrocatalysis applications using density function theory. Blue boxes indicate computational tasks which typically require researcher input. Factors above each task indicate potential acceleration through automation. Orange boxes are geometry optimizations via density functional theory calculations.

To best benchmark the traditional workflow against an automated one, we define the same objective for both paradigms: calculation of the adsorption energies of OH on the SAA of a Ni atom embedded on a Cu 111 surface, designated as Ni₁/Cu(111). This is further bounded to specifically include adsorption on all surface three-fold sites (6 in total). The goal is to mimic the scenario where an activity descriptor has already been identified for a specific electrochemical reaction, thereby collapsing performance predictions to the adsorption energy of a single adsorbate, as done previously [26]. As will be discussed later,

110 this represents an optimization problem of a binding energy across a set of possible SAAs
111 using SL to design the experiments. Previously, we have published methods to identify the
112 most robust descriptors for a given reaction based on uncertainty quantification techniques
113 [31, 32]. It should be noted that while automation generally replaces tasks that are on the
114 order of seconds and minutes, the accelerations reported from this category free up the user
115 to work on more analytical and constructive tasks, as elaborated in Section III.

116 All of the necessary steps to obtain the specified adsorption energies are highlighted
117 in Figure 2. A comparison of the estimated time required for each task in the traditional
118 approach and our automated approach is provided in Table I. Below, we outline the potential
119 acceleration for each task via automation.

120 1. *Candidate structure generation*

121 As an input, DFT requires atomic scale structural representations of the candidate sys-
122 tems to be evaluated. Structure generation in the context of electrocatalysis consists of
123 generation of the catalyst structure without any reaction intermediates, identification of all
124 of the possible adsorbate sites, and placement of the reaction intermediates on the sites of
125 interest. In this work we are not considering solvation effects. The first task corresponds
126 to writing and executing scripts to generate the clean Ni₁/Cu(111) slab via either ASE or
127 *AutoCat* (corresponding to the manual and automated approaches, respectively), and com-
128 paring the relative timings. While ASE has functions tailored for the generation of some
129 classes of systems, additional user involvement is necessary for those that are not currently
130 implemented. As an example, ASE does not currently have functions geared specifically
131 towards SAAs, and thus additional scripts are necessary to perform the doping of the pure
132 slabs. To generate each SAA the dopant site needs to be identified, the substitution made,
133 and spin polarization added to both the host and dopant as necessary. We can contrast
134 this with automation software such as *AutoCat* which has a function built on top of ASE
135 to streamline the generation of these SAA systems. Here, *AutoCat* can be viewed as fully
136 automating ASE towards a specific application (in this case, SAAs). Moreover, the code is
137 catered towards generating multiple SAAs through a single function call by the user that
138 writes to disk in an organized, predictable fashion. By leveraging tools for streamlined
139 candidate structure generation, a speedup of approximately 500× is observed. Thus, au-

140 tomatation of this task, while relatively straightforward in some cases, does present an avenue
141 for workflow acceleration.

142 Estimation of manual site identification for the second task of adsorbate placement re-
143 quires measuring the time it takes a graduate student team member to identify all of the
144 symmetrically unique surface sites of $\text{Ni}_1/\text{Cu}(111)$. This task becomes increasingly challeng-
145 ing from a user standpoint as the candidate catalyst becomes more elaborate, particularly
146 with broken surface symmetries. For example, in the case of SAAs, the presence of the
147 single-atom breaks many of the symmetries, and correctly identifying all unique sites by
148 hand is nontrivial. Some sites that would be regarded as symmetrically equivalent on a uni-
149 form surface can no longer be regarded as such due to the substitution of the single-atom. In
150 contrast, *AutoCat* identifies symmetry sites via the Delaunay Triangulation implementation
151 within the *pymatgen* software package [33], providing a systematic automated approach to
152 site identification that does not require user intervention. Comparison of the time required
153 for a graduate student team member to identify all of the sites relative to the automated
154 approach shows a speedup by a factor of $1000\times$. In summary, comparing the timings of
155 these three tasks (catalyst surface generation, site identification, and adsorbate placement)
156 highlights the effect of automation with regard to candidate structure generation tasks.

157 2. *Density functional theory pre- and post-processing*

158 For every catalyst structure generated, geometry optimizations via DFT calculations are
159 required. The total energies from these relaxed structures can then be used to estimate
160 properties of interest, such as adsorbate binding energy. Preparation for each of these
161 calculations involves writing DFT input files and scripts to submit these calculations to high-
162 performance computing (HPC) resources. The DFT input files contain all of the calculation
163 settings to be used, such as the k-mesh and exchange-correlation functional. In addition,
164 job submission scripts contain information about the requested computational resources
165 on a cluster, including the number of cores needed and the wall-time before the job will be
166 forcibly stopped. These scripts are necessary for every DFT calculation, and thus present an
167 opportunity for automation. To obtain a baseline, we time a user performing both the above
168 script writing tasks, i.e., generating DFT input files and batch submission scripts. This is
169 then compared to the time required for the equivalent tasks within our DFTitC framework.

170 We observe that these automated tasks are approximately $1000\times$ faster than their manual
171 equivalent, again emphasizing that the automation of these DFT pre-processing tasks is a
172 worthwhile effort.

173 Additionally, once the DFT calculations have successfully completed, the compilation of
174 results and data can consume a significant amount of time. The user must read through
175 each of the DFT output files, extract the desired information, and collect and organize this
176 data. Scaling up to a large number of systems, and thus calculation outputs, this can quickly
177 become a sizeable task. Here, we take timings of how long our DFTitC software takes to
178 parse the output data and compare it to the time taken to manually read all of the output
179 files and collect all of the data into a single spreadsheet. From automating this compilation
180 procedure we observe a speedup by a factor of $20\times$.

181 Given reference states, adsorption energies can be calculated from the total energies. We
182 thus compare the time required to calculate these adsorption energies within a spreadsheet to
183 that of automated calculations, which we observe to be $100\times$ faster when streamlined. This
184 final post-processing step of calculating the adsorption energies is relatively quick regardless
185 of the approach taken compared to the other steps considered in this workflow.

186 3. Workflow integration

187 In addition to the automation of structure creation and DFT pre- and post-processing
188 as described above, the automation of the submission of batch jobs to HPC clusters, status
189 monitoring, and general job management also provide opportunities for significant acceler-
190 ation. DFT calculations of catalyst structures are computationally expensive and typically
191 require active monitoring by the researcher. In particular, as these calculations can take
192 variable lengths of time, they may demand user intervention. For example, this could be to
193 fix errors or resubmit jobs, often at unpredictable times. This introduces “human lag” as it
194 is not possible for the typical researcher to continuously monitor the status of all submit-
195 ted DFT jobs. Here, human lag is modelled via a Monte Carlo sampling approach. First,
196 days are subdivided into three different windows representing typical working hours, hours
197 where some monitoring may occur, and hours where usually no monitoring would occur,
198 with “checkpoints” in time defined for each. Next, a uniform distribution is assumed for the
199 job finishing on any day of the week, without any preference for weekdays or weekends. Fi-

nally, we simulate the process of completion of a DFT job followed by research action at the nearest checkpoint in time, gathering statistics for a total of 10^6 DFT jobs, such as average lag per DFT job. In contrast, since job management within the fully-automated workflow is handled by a pipeline involving DFTitC and the **fireworks** [34] software package, there is no equivalent human lag, which enables significant acceleration.

Workflow step	Traditional	Automated	Acceleration
Catalyst structure generation			
Clean surface	16 min	2 s	~500x
Site identification	10 min	1 s	~1000x
Adsorbate placement	9 min	1 s	~1000x
DFT pre- and post-processing			
Generating DFT input and job management scripts	9 min	1 s	~1000x
Data collection	3 min	9 s	~20x
Adsorption energy calculation	2 min	1 s	~100x
DFT job submission and management			
Job resubmission and error handling	9 hr	–	–

TABLE I. Acceleration from automation of computational tasks and workflows.

B. Calculation Runtime Improvements

In the next category of acceleration, we quantify the speedup of calculation runtimes. Within our electrochemical materials discovery workflow, the primary physics-based simulation is DFT. As these calculations can be time-intensive, improving their runtimes is crucial in achieving significant acceleration.

In the case of adsorption structures, the initial guesses of the adsorbate geometry can play a key role. If the initial guess is far from the equilibrium geometry, more optimization steps will be required to achieve relaxation. Since each step requires a full DFT calculation to get the energy and forces, the initial guess should ideally be as close to the equilibrium

214 as possible to decrease the overall calculation runtime. The total runtimes of geometry
 215 optimizations via DFT can also be heavily influenced by the choice of calculator settings,
 216 such as initial magnetic moment. Starting with a poor guess of the initial magnetic moment
 217 could result in longer time to achieve convergence in the DFT algorithm. To decouple the
 218 influences of both initial geometry guess and appropriate choice of calculator settings, we
 219 run four sets of relaxations for OH on all of the hollow sites of Ni₁/Cu(111). We use two
 220 initial geometry guesses, one of which we call a chemically informed configuration as it
 221 is relatively close to the relaxed configuration. The initial height for this configuration is
 222 guessed based upon the covalent radii of the nearest neighbors of the anchoring O atom. For
 223 comparison, a chemically naive configuration is also considered where the initial geometry
 224 is further from the relaxed configuration, with the OH bond angle at 45° with the surface.
 225 Moreover, the initial height in this case is taken to be 1.5 Å above the surface. In addition to
 226 the different geometries, we also take two approaches to setting the initial magnetic moment
 227 of the single-atom. One approach is to choose the initial magnetic moment based on the
 228 ground-state magnetic moments of the single-atom species from ASE. This approach tailors
 229 the selection of initial magnetic moment to each particular system. Alternatively, we also
 230 test applying a default starting initial magnetic moment of 5.0 for the dopant, regardless
 231 of the identity of the single-atom species. This is a relatively naive approach as it does
 232 not incorporate details of the specific system for this choice of setting. In this specific case
 233 of Ni₁/Cu111, since the structure prefers to be in a spin-paired state (i.e., without spin-
 234 polarization), the former approach provides a guess closer to the actual spin-polarization for
 235 which the system converges. Our intention here is to highlight the impact of these aspects
 236 of the DFT calculations on acceleration and could stem from deterministic algorithms, an
 237 ML model, or another approach entirely.

238 In Figure 3 we visualize the accelerations of the DFT runtimes from both the calculation
 239 settings and initial geometries. Firstly, we observe relatively modest speedups from choice
 240 of calculator settings with average values of approximately 1.1× for both the naive and
 241 informed geometries. For these calculations, the system converges to the non-polarized
 242 state within the first few steps. Thus the observed speedup from choice of initial magnetic
 243 moment of the single-atom is mainly a reflection of these initial iterations while it finds
 244 the appropriate spin state, which often take the longest. On the other hand, we observe a
 245 larger acceleration effect from the initial geometry. When keeping the settings to be chosen

naively or tailored while changing the initial geometry, we observe a speedup of $2.1\times$ and $2.2\times$, respectively. The speedup described here can be mainly explained by the decrease in the number of steps required to reach the equilibrium configuration for a fixed optimization scheme. Using the tailored calculation settings, an average of approximately 33 and 16 geometry optimization steps with the Broyden-Fletcher-Goldfarb-Shanno (BFGS) algorithm are taken when starting from the chemically naive and informed geometries, respectively. Similarly, when using the naive calculation settings the average number of optimization steps for the chemically naive geometry is 34 and for the chemically informed geometry it is 17. Thus, methods to decrease the number of steps taken to reach equilibrium and shorten the DFT compute time at each step are highly desirable, and are an area of active development [35–40]. Combining both the improved initial geometry as well as the choice of calculator settings yields the largest runtime acceleration observed in this work of $2.3\times$, thus motivating the consideration of both variables within automated workflows. As will be discussed in greater detail in section II E, DFT takes up a substantial portion of the overall pipeline runtime, and thus this acceleration factor is indicative of potentially enormous improvements in overall acceleration.

Workflow step	Traditional	Automated	Acceleration
DFT calculation settings and initial structure guess			
Clean substrate relaxation	21 hr	18.5 hr	$\sim 1.1\times$
Substrate + adsorbate relaxation	46 hr	20 hr	$\sim 2.3\times$

TABLE II. Acceleration from calculation runtime improvements.

C. Efficient Design Space Search

Next, we estimate the acceleration resulting from use of a SL workflow for selecting and evaluating candidates in a design space of catalysts and compare it to that of traditional approaches. The SL workflow proceeds as follows: (1) start with an initial set of a small number of training examples of catalyst candidates and their properties; (2) build ML models using the initial set of training examples and predict the objective properties of all the candidates in the design space of interest; (3) use an acquisition function that considers

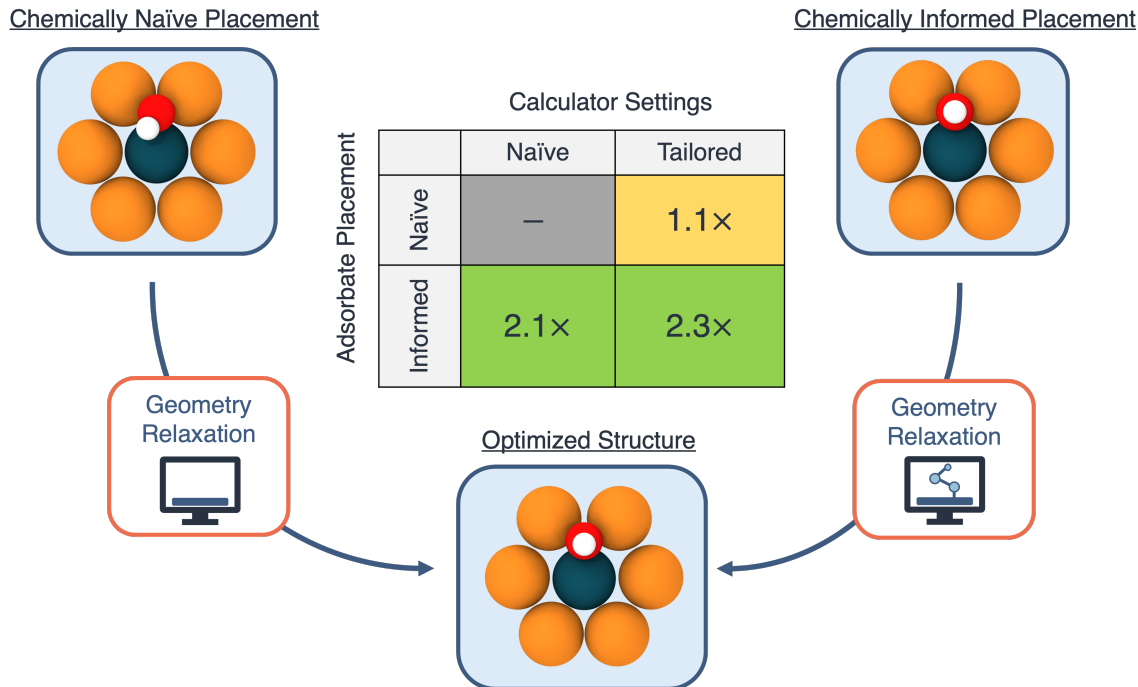


FIG. 3. Estimated density functional theory geometry optimization runtime accelerations. These are decoupled between chemically informed or naïve initial geometries, and tailored or naïve calculation settings. The largest factor of acceleration is observed when using an informed structure generation with tailored calculator settings.

model predictions and uncertainties to select the next candidate to evaluate; (4) evaluate the selected candidate and add it with its newly obtained label to the training set; (5) iterate steps 2–4 in a closed-loop manner until a candidate, or a certain number of candidates, with the target properties has been discovered. A detailed schematic of this workflow is presented in Figure 4. Such a strategy has been previously shown to be more efficient in sampling the design space to find novel candidates by a factor of 2 – 6× over traditional grid-based searches or random selection of candidates from the design space [9–25].

For benchmarking the acceleration from SL for a typical catalyst discovery problem, we use a dataset of ~300 bimetallic catalysts for CO₂ reduction [41]. The dataset contains ~30 candidates with the target property of *CO adsorption energy on the catalyst surface inside a narrow window of [−0.7 eV, −0.5 eV]. We perform an SL simulation, starting with a small initial training set of 10 examples from the above dataset, and iterate in a closed-loop as described above until all the target candidates in the design space have been identified

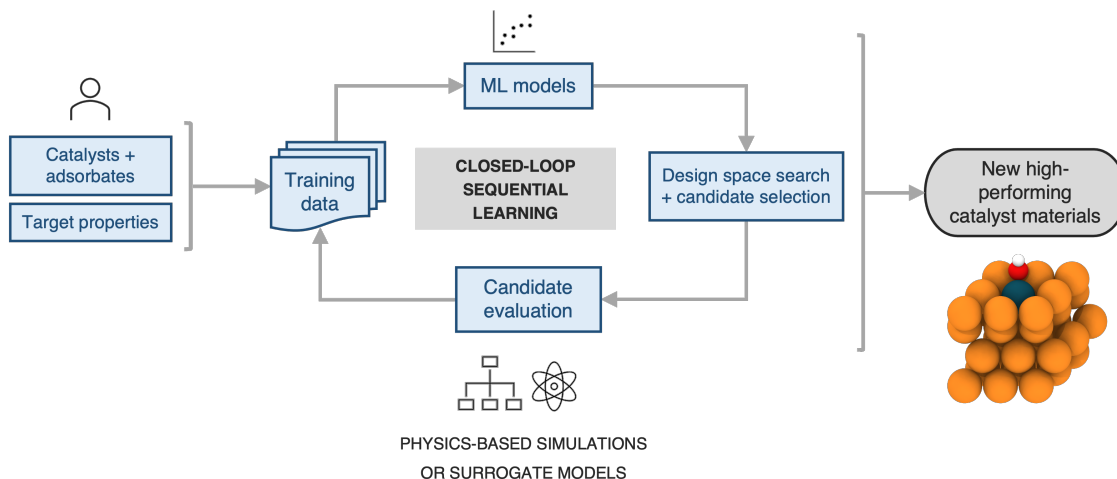


FIG. 4. A typical closed-loop sequential learning workflow for computational discovery of novel catalyst materials.

successfully, and benchmark the acceleration against random search. In particular, at each SL iteration, we build random forest-based models using the *lolo* software package [19], and predict the $^*\text{CO}$ adsorption energies of all candidates along with robust estimates of uncertainty in each prediction. The next candidate to evaluate is chosen based on the maximum likelihood of improvement (MLI) acquisition function. This function selects the system with the maximum likelihood of having an adsorption energy in the $[-0.7 \text{ eV}, -0.5 \text{ eV}]$ window, when considering both the predicted value as well as its uncertainty. Overall we find that such an SL-based workflow successfully identifies all ~ 30 target candidates $3\times$ faster than random search (Figure 5a). In addition, we note that the candidates surfaced by SL, on average, have properties closer to the target property window than those surfaced by random search, even when those candidates do not explicitly fall within the window (Figure 5b). In other words, in addition to discovering target candidates considerably more efficiently than random search, an SL-based approach surfaces potentially interesting candidates near the target window much more frequently than random search.

D. Surrogatization of Compute-Intensive Simulations

For the last category of acceleration, we estimate the extent of further possible speedup through the surrogatization of the most time-consuming tasks in the workflow. In particular,

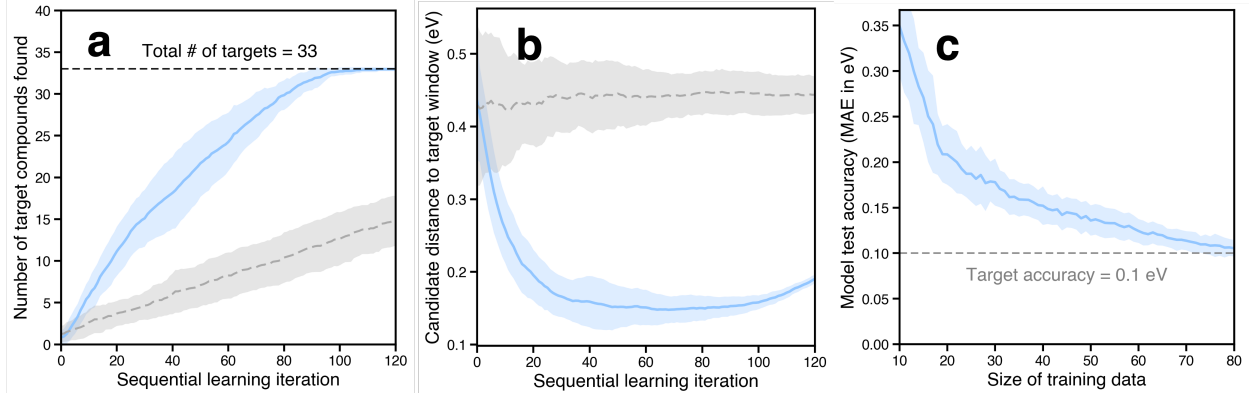


FIG. 5. Comparison of random search vs sequential learning (SL)-driven approach to find new bimetallic catalysts with a target property. (a) Overall, the SL-driven approach identifies all the 33 target candidates in the dataset within 100 iterations, $\sim 3\times$ faster than random search. (b) Candidates surfaced via SL lie much closer to the target window on average, when compared to those identified via random searching. (c) An SL-driven approach can help identify a much smaller number of examples that can be used to train ML surrogates to a desired accuracy, at a fraction of the overall dataset size. Here, the overall dataset has ~ 300 candidates, and an ML model trained on only $\sim 25\%$ of the candidates chosen via a SL-driven maximum uncertainty-based approach achieves the target accuracy.

the rate determining step of the closed-loop framework considered here is the calculation of the binding energies of adsorbates using DFT. ML models can be used as surrogates for physics-based simulations of material properties often at a fraction of the compute cost and with marginal loss in accuracy. The primary cost of building such ML surrogates for materials properties often lies in the generation of training data where such data does not exist, especially when the data generation involves compute-intensive physics-based simulations such as DFT. Here we estimate the size of such training data required to build and train ML surrogates with a target accuracy, especially when such training data is iteratively built using an SL-based strategy.

We use the dataset of bimetallic catalysts for CO_2 reduction mentioned in Section II C within a SL workflow to simulate an efficient, targeted training set generation scheme. Similar to the SL workflow employed in the search for novel catalyst materials in a design space of interest, we employ a closed-loop iterative approach to generate the training data and address model uncertainty. We start with a small initial training dataset of 10 systems,

313 build random forest models to predict adsorption energies, and iteratively choose the next
 314 candidate to build the training data. With model accuracy in mind, we employ an acquisi-
 315 tion strategy that optimizes for the most accurate ML model on average. In particular, at
 316 each iteration the candidate whose property prediction has the maximum uncertainty (MU)
 317 is selected to augment the training data. Inclusion of such a candidate results in the high-
 318 est improvement of the overall accuracy of the ML model by targeting areas of the design
 319 space which are not well reproduced by the model. Using an accuracy threshold of interest,
 320 we then determine the fraction of the overall training data necessary for building useful ML
 321 surrogates. For instance, with a threshold of 0.1 eV (the typical difference between DFT and
 322 experimental formation energy values [42]), we estimate that accurate ML surrogates can be
 323 trained using a dataset generated via the above SL-strategy with $\sim 25\%$ of the overall dataset
 324 size (Figure 5c). The accuracy metric here is calculated via a bootstrapping approach for
 325 the test set, with additional details provided in the Supplementary Information.

326 **E. Overall Acceleration of the Full End-to-end Workflow**

327 Finally, we aggregate the acceleration from the various steps in the workflow to estimate
 328 the overall speedup achieved. Here, we use the single-atom alloys (SAA) design space for
 329 calculating the overall estimates. We begin by estimating the size of such a design space.
 330 Limiting the design space to ~ 30 transition metal hosts and dopants results in a total of
 331 $\sim {}^{30}\text{C}_2 \approx 900$ SAA systems. For each SAA system, typically a few (3–5) low-index surface
 332 terminations are considered. Moreover, the considered reaction intermediate can adsorb
 333 onto the catalyst surface at one of many possible symmetrically unique sites (up to 20–40
 334 configurations), and all such possible intermediate configurations need to be considered in
 335 the design space. Overall, a typical SAA design space when fully enumerated can have up
 336 to 10^5 – 10^6 possibilities.

337 Using the above SAA design space, we apply the estimated time for each step in our overall
 338 end-to-end catalyst workflow as derived in the previous sections, using both traditional and
 339 automated methods (with and without surrogates), and calculate the overall speedup. From
 340 the automation of tasks and workflows, and runtime improvements alone, we achieve an
 341 acceleration of $\sim 10\times$ (a reduction of $\sim 90\%$) over traditional materials design workflows.
 342 Further utilizing the ML surrogates (including the compute costs required to generate the

training data) can result in an acceleration of up to $\sim 25\times$ (a reduction of up to $\sim 96\%$) over traditional approaches.

Approach	Structure Generation	Substrate Calculation	Adsorbate Placement	Catalyst Calculation	Data Usage	Post-processing	Design Space Search Factor	Total Acceleration
Traditional	16 min	21 hr	18 min	72 hr/i.c. ¹	100%	5 min	1	
Automated	2 s	18.5 hr	2 s	20 hr/i.c. ¹	100%	10 s	0.33	10 \times
+ Surrogates	2 s	—	2 s	20 hr/i.c. ¹	10-25% ²	2 s	0.33	15-20 \times

TABLE III. Overall acceleration benchmarks for the end-to-end workflows with and without surrogatization. We demonstrate a speedup of up to 10x with automation of tasks and runtime improvements, and a speedup of up to 25x upon using ML surrogates for the most compute-intensive DFT tasks. ¹i.c. = intermediate configuration (total # i.c. \approx 200/catalyst system); “traditional” includes human lag estimates. ²estimate from bimetallic catalyst dataset of the relative amount of DFT data needed to reach a target accuracy of 0.1 eV/adsorbate.

III. DISCUSSION

The results presented here have implications that reach beyond the reported factors of acceleration. Here, it is helpful to make a distinction between project time and researcher time. We consider project time as the time necessary to carry a project to completion. In other words, this is an accumulation of all the time spent towards achieving the tasks to reach the project goal. Thus, all the acceleration factors quantified above are with respect to this project time. Therefore, these workflows are anticipated to have a direct impact on project time to completion. In addition, by breaking down the acceleration benefits for each component of the workflows, estimates of project time acceleration for differing framework topologies (e.g. multi-scale evaluation) than those outlined here can be inferred.

On the other hand, researcher time can be interpreted as time spent from the frame of reference of the researcher on a given workday. The acceleration associated here is not directly quantified as we have done with project time. Instead, this acceleration is an indirect consequence of implementing these workflows. The most obvious example of this influence is through task automation. In the traditional paradigm, these tasks can become time-

360 consuming, particularly as the scale of the project increases. Automation frees up valuable
361 researcher time that would normally be occupied by the more mundane tasks. This allows
362 the researcher to instead focus on more intellectually demanding tasks such as performing
363 literature surveys and project formulation, improving research productivity.

364 Automating job management has the benefit of impacting both project time and re-
365 searcher time. Since this form of automation facilitates running computational jobs around-
366 the-clock, the human-lag as described by our model is entirely removed. If a job were to
367 finish outside of working hours, there will not be any lag. This decreases the project time
368 as described above. In the context of researcher time, this automation also has the added
369 benefit of decreasing the need for regular job monitoring. Thus, during the day a researcher
370 can devote more time towards other tasks.

371 A few additional observations regarding the nature of the baselines used to estimate the
372 speed of traditional approaches in this work are warranted. First, for estimation of task
373 timings such as input file generation for simulations and script generation to submit jobs on
374 HPC resources, we use time estimates from a single researcher. The timings of such tasks
375 are inherently variable, depending on the exact nature of the task, the researcher performing
376 it, as well as the environmental setup in which it is performed. Similarly, natural delays as-
377 sociated with monitoring and managing ongoing computational jobs depend on the working
378 habits of the researcher, the time-scale associated with each computation (e.g., those that
379 take hours opposed to days or weeks to complete), and the availability or connectivity of the
380 computational resources (e.g., on-site resources versus those that can be accessed remotely).
381 Lastly, to estimate the acceleration from an intelligent exploration of the design space using
382 sequential learning, we use random sampling as the benchmark. While random sampling is
383 an excellent exploratory acquisition function [43] [include REF, e.g., several hyperparameter
384 tuning algorithms use random search over a grid to find the optimal set within a fraction
385 of full grid search: <https://www.jmlr.org/papers/volume13/bergstra12a/bergstra12a.pdf>.
386 Bryce may have a reference handy that is more directly relevant for SL/design of exper-
387 iments], it is not a substitute for traditional methods of design space exploration. Typically,
388 traditional search approaches are influenced by prior knowledge, research directions within
389 the community at the time, available resources, among other factors. We use random search
390 here, not least because a model to predict a traditional materials design trajectory does
391 not exist, to our knowledge, but also because it is widely-used as an unbiased exploratory

baseline [9–13, 15–17, 19–25].

We want to emphasize that, given some of the variability in the baselines as discussed above, the goal of this work is to highlight the scale of acceleration that can be attributed to the several individual components in a computational materials design workflow. Moreover, we also aim to highlight the challenges associated with estimating such factors of acceleration, and not necessarily the raw factors themselves. Our work underscores the benefits of data collection and sharing, especially around time spent on research tasks, monitoring and managing medium- to high-throughput computational projects, implementing traditional approaches of materials discovery and design trajectories, and handling failed computations and experiments. We recommend a community-driven initiative towards such data collection and sharing efforts to bolster our understanding of the traditional baselines as well as to further contextualize the significant benefits of automation and ML-guided strategies.

IV. CONCLUSION

In this work we demonstrate that software automation and runtime improvements combined with a sequential learning-based closed-loop search over a design space for new catalysts can provide an overall acceleration of more than 10x (or more than 90% reduction in overall time/cost) over traditional approaches. Further, we estimate that such automation frameworks can have a significant impact on researcher productivity (20–1000×), direct compute costs (1.1–2.3×), and project/calendar time (>10–20×). This was estimated using 3 sources of acceleration. Through combination of manual computational experiments with timing of automated equivalent tasks, we provide speedup estimates stemming from each category. Automation of tasks can provide improved discovery time by streamlining tasks that usually need to be completed via user intervention. We also identify that significant speedup in terms of DFT runtime can be achieved through better initial prediction of the catalyst geometries and calculator settings. Moreover, the use of a sequential learning framework to guide design of experiments can accelerate discovery by a factor of 3, thereby dramatically decreasing the number of full loop iterations that need to be performed to reach a given design goal. We extend this analysis to include replacement of DFT calculations with machine-learning surrogates, another source of acceleration, and observe that this discovery speedup factor can be further improved to >15–20×. We believe that the results outline the

immense benefits of introducing automation and machine learning into scientific discovery workflows, and motivate the increasingly widespread adoption of these methods.

V. METHODS

A. Workflow Topology

In this work we consider two different closed-loop topologies. The first is a two-stage process consisting of DFT calculations to calculate adsorption energies which are then fed into a SL model to iteratively guide candidate selection (Figure 1a). Within the DFT stage, multiple steps need to be taken for study of an electrocatalysis problem. Namely, a geometry relaxation of the clean candidate system, followed by a relaxation of all reaction intermediates onto the unique surface symmetry sites. These DFT calculations are taken to be automated within a pipeline. Here we use a combination of *AutoCat* (https://github.com/aced-differentiate/auto_cat) for automated structure generation and the *DFTitC* ecosystem for the calculations themselves. More details on these software packages are provided in Section V B. The SL stage then serves to guide the design space search by iteratively identifying candidates to evaluate on each loop. Additional details on the models used for this purpose are described in Section V D.

Another topology we consider is an extension of that described above with ML surrogates for the DFT calculations introduced into the loop (Figure 1b). In this scenario, the first few overall iterations proceed the same as before, except now as the DFT data is generated, a surrogate ML model is trained on the resulting data until a threshold test accuracy is reached. For these first number of iterations, candidates are selected with the intent of improving prediction accuracy. Once the threshold accuracy for the surrogate is met, all subsequent iterations of the loop will instead use the surrogate to obtain adsorption energies. From this point onward candidate selection is then focused on identifying promising materials using this trained ML model for continued exploration, as before in the loop without surrogatization. Motivation for such an approach stems from the computationally demanding nature of DFT calculations which is an inevitable bottleneck of these calculations. Swapping these heavy calculations for an essentially zero-cost alternative to obtain the same results is desirable in terms of overall speed. If desired, candidates identified via the ML model surrogate can be

451 validated using DFT, but we are not including this aspect explicitly in our analysis. Thus,
452 in this extended workflow we place emphasis on the number of DFT calculations necessary
453 to obtain a surrogate meeting the minimum testing accuracy.

454 B. Automation Software

455 To create the atomic structures for the DFT calculations, we use *AutoCat*, a software
456 package with tools for structure generation and sequential learning for electrocatalysis ap-
457 plications. This package is built on top of the Atomic Simulation Environment (ASE) [29]
458 and `pymatgen` [33] to generate the atomic structures *en masse*, and write them to disk follow-
459 ing an organized directory structure. To generate single-atom alloys, *AutoCat* has tailored
460 functions for this purpose with optional selection of supercell dimensions, vacuum spacing,
461 as well as number of bottom layers to be fixed, with appropriate defaults for each parameter.
462 Moreover, through the use of `pymatgen`’s implementation of Delaunay triangulation [44], all
463 of the unique symmetry sites on an arbitrary surface can be identified. Furthermore, initial
464 heights of adsorbates are estimated through the covalent radii of the anchoring atom within
465 a given adsorbate as well as its nearest neighbors. As the development of this package is
466 part of an ongoing work, additional details will be reported in a future publication.

467 Once the catalyst with adsorbate systems have been generated by *AutoCat*, the crystal
468 structures are used as input to an automated DFT pipeline that (a) generates input files
469 for a DFT calculator (here we use GPAW[45, 46]), (b) executes DFT calculation workflows,
470 and (c) parses successfully completed calculations and extracts useful information.

471 *Automatic DFT input generation:* We leverage the Python-based *dftinputgen* package
472 (<https://github.com/CitrineInformatics/dft-input-gen>) to automate the generation
473 of DFT input files from a specified catalyst/adsorbate crystal structure. In particular, we
474 extend the *dftinputgen* package to support GPAW. For a given input crystal structure, the
475 package provides sensible defaults to use for commonly-used DFT parameters based on prior
476 domain knowledge for novice users as well as fine-grained control over each parameter for
477 more experienced DFT practitioners. The package also implements, “recipes”, sets of DFT
478 parameters and values to be used as default depending on the properties of interest, e.g.,
479 ground-state geometry and electronic structure. The package outputs input files in a user-
480 specified location that can be directly used by mature DFT packages as input for calculation.

481

482 *Execution of DFT calculation workflows:* We leverage the Python-based *fireworks* [34] pack-
 483 age to both define complex sequences of DFT subcalculations necessary for electrocatalysis
 484 studies (e.g. clean surface relaxation followed by adsorption relaxation), and to create,
 485 submit, and monitor batch jobs on HPC resources that correspond to such a sequence of
 486 subcalculations. These scripts are part of an ongoing study and will be open-sourced.

487 *Parsing output from DFT:* After having completed DFT calculations on a large number of
 488 different candidate systems, key metrics such as total energy and forces would need to be ex-
 489 tracted in bulk. To accomplish this task we have extended the previously-developed *pif-dft*
 490 (<https://github.com/CitrineInformatics/pif-dft>) and *dftparse* ([https://github.](https://github.com/CitrineInformatics/dftparse)
 491 [com/CitrineInformatics/dftparse](https://github.com/CitrineInformatics/dftparse)) packages to parse output generated via GPAW. Func-
 492 tions written for this package can look for a *.traj* file resulting from a successful GPAW
 493 run, in a specified directory. Once a traj file has been identified, it can be read using
 494 ASE to extract calculated properties of interest. This includes not only results such as
 495 total energy and forces, but also calculator settings such as the exchange-correlation im-
 496 plemented. These findings are then written into a Physical Information File (PIF) [47]
 497 (<https://citrine.io/pif>), a general-purpose materials data schema, for every calculation
 498 conducted.

499 C. First-Principles Calculations

500 All DFT calculations are conducted with the GPAW package [45, 46] via ASE [29]. The
 501 projector-augmented wave method is used for the interaction of the valence electrons with
 502 the ion cores. A target spacing of 0.16 Å is applied for the real-space grid, with a Monkhorst-
 503 Pack [48] k-mesh of $4 \times 4 \times 1$ for all surface calculations. For improved self-consistent field
 504 convergence, a smearing width of 0.05 eV is applied through the Fermi-Dirac distribution.

505 For the computational experiments adsorbing OH onto the Ni₁/Cu(111) hollow sites
 506 described in Section II B, we employ two approaches for both the selection of calculator
 507 settings as well as initial geometry configuration. With regard to the calculator setting
 508 approaches, the tailored approach gives the dopant an initial magnetic moment based on
 509 the ground-state magnetic moment from ASE. On the other hand, the naive approach to
 510 calculator settings, where a uniform value is given regardless of species (in this case 5.0),

511 is used for the dopant initial magnetic moment. In terms of the initial geometries, the
512 chemically naive approach places OH at 45° with respect to the surface and an initial height
513 of 1.5 \AA above the surface. In contrast, the chemically informed initial structure has the OH
514 bond as perpendicular to the surface plane. Moreover, the initial height for the chemically
515 informed approach is an average based on the covalent radii of the nearest neighbor species
516 at a given xy coordinate. All geometry optimizations are conducted via the BFGS algorithm
517 as implemented in ASE.

518 D. Machine Learning Models

519 All ML models reported in this work are based on random forests [49], consistent with
520 the previously-reported FUELS framework [19] and as implemented in the open-source *lolo*
521 library [50]. Materials in the training dataset are transformed into the Magpie features [13],
522 a set of descriptors generated using only the material composition, as implemented in the
523 *matminer* package [51]. The uncertainty in a model prediction is determined using jackknife-
524 after-bootstrap and infinitesimal jackknife variance estimators [52].

525 CONFLICTS OF INTEREST

526 There are no conflicts to declare.

527 ACKNOWLEDGEMENTS

528 The work presented here was funded in part by the Advanced Research Projects Agency-
529 Energy (ARPA-E), U.S. Department of Energy, under Award Number DE-AR0001211. L.K.
530 acknowledges the support of the Natural Sciences and Engineering Research Council of
531 Canada (NSERC). The authors thank Rachel Kurchin for helpful discussions around au-
532 tomation and acceleration estimation.

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