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Critical currents of superconducting aluminium - germanium and lead - germanium thin film alloys near the metal - insulator transition (*)

G. Deutscher and M. L. Rappaport

Department of Physics and Astronomy, Tel-Aviv University, Tel-Aviv, Israel

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Résumé. — Les courants critiques de films minces d'alliages Al-Ge et Pb-Ge ont été mesurés en fonction de la concentration en volume de métal x . Nous avons trouvé qu'ils varient comme $(x - x_c)^v$ où x_c est la valeur de x à la transition métal-insolant. Ce comportement est similaire à celui de la conductance σ des mêmes échantillons à l'état normal. Nous interprétons ces résultats dans le cadre des théories récentes du phénomène de percolation.

Abstract. — The critical currents of Al-Ge and Pb-Ge thin film alloys have been measured as a function of the metal volume fraction x . They have been found to vary as $(x - x_c)^v$ where x_c is the value of x at the metal-insulator transition. This behaviour is similar to that of the conductances σ of the same samples in the normal state. We interpret these results in terms of current percolation theories.

We report in this letter the first critical current measurements on superconductor-semiconductor mixtures as a function of the metal volume fraction x near the metal-insulator threshold x_c , and we propose an interpretation of the results in terms of the percolation model of de Gennes [1] and Skal and Shklovskii [2]. This interpretation leads to the conclusion that the critical current $j_c \propto (x - x_c)^v$ with $v = (d - 1) \nu$ where d is the dimensionality of the sample and ν the critical index for the coherence length relevant to the percolation problem. In the model of references [1] and [2] the critical index t for the normal state conductance σ is given by $t = (d - 1) \nu + \delta$ where δ is a *twistedness* index for the percolative paths ($\delta > 0$). Experimentally we find within our experimental uncertainty that $v = t = 1.75 \pm 0.15$ in 3D Al-Ge films and we conclude that the twistedness of the percolative paths does not actually decrease the conductance of the network, which is in practice governed (like the critical current density in the superconducting state) only by the density of percolative paths. In the (probably) 2D Pb-Ge films,

$$v = 1.3 \pm 0.1 \quad \text{and} \quad t = 0.9 \pm 0.1 :$$

it appears that in this regime the model of Skal and Shklovskii is not valid since $\delta < 0$.

The electrical properties of metal-insulator mixtures have recently been the subject of considerable attention [3], [4]. Measurements have primarily concentrated on the rapid change of σ near the metal-insulator transition. In accordance with percolation theory [5] it has been observed that $\sigma \propto (x - x_c)^t$ where t falls in general in the range from 1.5 to 1.9, both experimentally and from theoretical estimates. However, the exponent t is not directly related to other exponents of the percolation problem for which renormalization group theories make accurate predictions. Moreover, experimental determinations of t suffer from the fact that σ does not actually go to zero at $x = x_c$ due to the finite conductivity of the matrix and to quantum effects [6].

The advantages of critical current measurements are that they are related to the better known exponent ν and that at concentrations $x < x_c$ the critical current is zero. We have performed measurements on Pb-Ge ($x_c = 18\%$) and Al-Ge ($x_c = 55\%$) thin film alloys, where the former system has a random structure and a value of x_c in accordance with that predicted by percolation theory in a random continuous medium while the latter has a highly regular granular structure [7] and a value of x_c close to that of a close packed array of spheres.

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Samples were prepared by evaporation of the constituents from two electron beam guns onto room temperature glass substrates at a pressure of 5×10^{-6} torr during evaporation. The evaporation rate from each gun was monitored and controlled through a separate quartz thickness monitor. Evaporation rates were held constant to within a few percent. Nine samples of thickness close to 2000 \AA were deposited simultaneously on each substrate, sample composition varying across the substrate due to varying distances from the evaporation sources. Sample concentrations were established using the quartz thickness monitor readings and deposition profiles measured for each material by interferometry. Concentration difference between neighbouring samples (8.5 mm apart) was from one to two percent, depending on the particular concentration under investigation. This set up allows a fairly good knowledge of the concentration differences between samples deposited on the same substrate, while the absolute value of the concentration is only known to within several percent.

Normal state conductivities σ , critical temperatures T_c , and critical current densities j_c were measured on the same samples by the standard four point contact method. Critical current densities were determined as that producing a $1 \mu\text{V}$ potential difference across the sample (1 mm in length). In order to study concentration dependences near the metal insulator transition we used samples evaporated on a single substrate with a median concentration slightly above the threshold x_c . The value of x_c was chosen as that for which j_c went to zero. For Pb-Ge ($T_c \approx 7.2 \text{ K}$) j_c was measured at 4.2 K and for Al-Ge ($T_c \approx 1.8 \text{ K}$) at 1.42 K .

The results obtained for Al-Ge are presented in the form of a log-log plot in figure 1, which shows that both σ and j_c vary as $(x - x_c)^{1.75 \pm 0.15}$. The value obtained for the exponent depends somewhat upon the exact choice made for x_c , but this does not affect the main conclusion that the behaviour of j_c and σ near x_c are essentially the same. The value of 1.75 is in good agreement with that predicted for t by percolation theory for a 3D medium [8] (actually the Al-Ge films contain about 20 layers of Al grains so that the crossover from 3D to 2D would occur only very close to x_c). The departure from linearity observed for $\log \sigma$ at low values of $\log (x - x_c)$ may be due to the finite conduction of the Ge matrix and/or to quantum effects [6]. In Pb-Ge (Fig. 2) the critical index v is somewhat smaller than in Al-Ge (1.3 ± 0.1), possibly due to the fact that the film thickness is only about 4 times the grain size so that the films are 2D rather than 3D in the investigated range $x - x_c < 0.1$. Moreover, it is seen in figure 2 that $t < v$. The slope of the conductance data (0.9 ± 0.1) may be misleading due to the contribution of the Ge near x_c , but is consistent with the calculated value of 1.1 in 2D [8].

We now turn our attention to the comparison between the behaviour of j_c and that of σ , which we

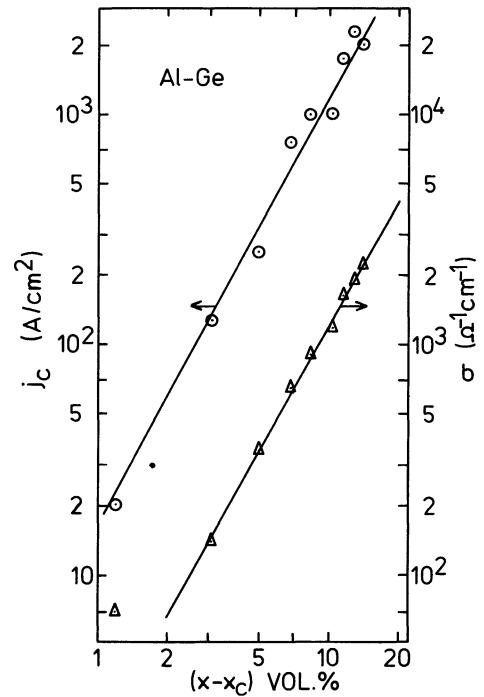


Fig. 1. — Log-log plot of the critical current density j_c and normal state conductance σ versus concentration in 3D Al-Ge, showing straight lines with slopes equal to 1.75 ± 0.15 .

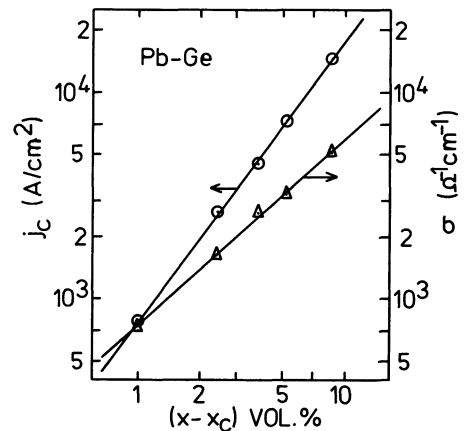


Fig. 2. — Log-log plot of the critical density $j_c \propto (x - x_c)^v$ and normal state conductance $\sigma \propto (x - x_c)^t$ in 2D Pb-Ge showing that $v (= 1.3 \pm 0.1) > t (= 0.9 \pm 0.1)$.

propose to interpret with the percolation model of de Gennes [1] and Skal and Shklovskii [2]. In this model, the current flows along macrobonds. The average distance between nodes is seen as the correlation length ξ of the percolation problem and varies as $(x - x_c)^{-v}$ while the average length of a macrobond between neighbouring nodes L varies as $(x - x_c)^{-v-\delta}$ where $\delta > 0$. The conductivity varies as $L^{-1} \xi^{2-d}$ and therefore its critical index

$$t = (d - 1)v + \delta$$

depends both on the density of the macrobond network and on the *twistedness* index δ .

The critical current is defined experimentally by the appearance of a certain flux flow voltage across the sample. The flux flow voltage that appears between two nodes due to flux creep through the intergrain junctions can be written as

$$V \propto NrI \exp - \frac{E_c - \frac{\hbar}{2e} I}{kT} \quad (1)$$

where N is the number of intergrain junctions along the macrobond, r the normal state resistance of one junction and E_c its Josephson coupling energy. At low temperatures $kT < E_c$ the critical current I_c (the value of the current at which a certain flux flow voltage V_0 appears) is essentially determined by E_c and depends only logarithmically on V_0 and the macrobond length. In this regime the critical current density of the sample is $j_c = I_c \xi^{1-d}$ so that $v = (d - 1) \nu$.

Therefore a comparison between the values of t and ν measured on the *same* set of samples allows a determination of the twistedness index δ . Our measurements on the 3D Al-Ge alloys give

$$t = 2 \nu = 1.75 \pm 0.15$$

in excellent agreement with the contention of Skal and Shklovskii [2] that $\delta \ll 1$ and with numerical estimates of t and ν [8]. The critical current data obtained for Pb-Ge — if we interpret them as characteristic of a

2D layer due to the rather small thickness to grain size ratio (~ 4) — gives $\nu = \nu = 1.3 \pm 0.1$, again in excellent agreement with calculated values [8]. The conductance data give — with the reservation mentioned above — a value of $t = 0.9 \pm 0.1$ which is smaller than ν . Such a result seems to invalidate the Skal and Shklovskii picture for $d = 2$. For that case we note that calculated values of t are actually smaller than ν [8].

In summary, by a comparison of conductance and critical current measurements on the same set of samples we have established experimentally for the first time that the twistedness of the percolation paths does not play an important role in the conductance of the network, as first suggested by Skal and Shklovskii. From the critical current data we have obtained values for the critical index ν in 3D (0.87 ± 0.08) and in 2D (1.3 ± 0.1) that are in good agreement with calculated values [8]. And finally, in 2D the slower variation of the conductance as compared to the critical current (corresponding to a negative value of the twistedness index δ) seems to invalidate the model of Skal and Shklovskii for that dimensionality, a point already mentioned by Kirkpatrick [8] on the basis of computer simulations.

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