

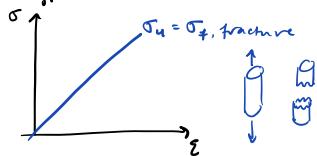
## Stress-Strain Mechanisms

$$\sigma = \frac{P}{A_i} \quad \text{engineering stress, initial, undeformed A}$$

$$\epsilon = \frac{\Delta L}{L_i} \quad \text{engineering strain, initial, undeformed A}$$

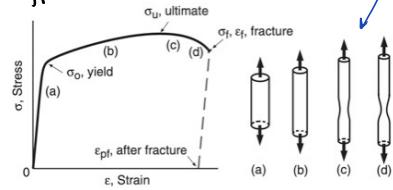
↑ sometimes given as percentage where  
 $\epsilon\% = 100\epsilon$   
 (microstrain  $\epsilon_m = 10^6\epsilon$ )

Typical Ceramic

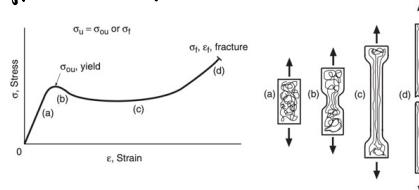


$$\text{Young's Modulus, } E = \frac{\sigma_B - \sigma_A}{\epsilon_B - \epsilon_A} \quad \left\{ \begin{array}{l} \text{tangent modulus may} \\ \text{be used if no well-defined linear region} \end{array} \right\}$$

Typical Ductile Metal



Typical Polymer



$\sigma_u$  = ultimate tensile strength, or just tensile strength  
 - highest engineering stress reached prior to fracture

$$\sigma_u = \frac{P_{\max}}{A_i}$$

$\sigma_f$  = engineering fracture strength  
 - obtained at point of fracture, even if not highest force reached

$$\sigma_f = \frac{P_f}{A_i}$$

Brittle materials:  $\sigma_u = \sigma_f$

Ductile Materials:  $\sigma_u > \sigma_f$

$\sigma_0$  = offset yield strength  
 - departure from linear-elastic behavior

Measures of Ductility - ability of a material to accommodate inelastic deformation without breaking

$\epsilon_f$  = engineering strain at fracture, usually expressed as percentage and termed:  
 $100\epsilon_f$  = percent elongation at fracture

$100\epsilon_{pf}$  = percent elongation after fracture

$$\epsilon_{pf} = \frac{L_f - L_i}{L_i} \quad \text{where } L_f = \text{fracture length, } L_i = \text{distance between tick marks before testing}$$

Elastic strain is lost when the stress drops to zero after fracture, so this quantity is a plastic strain

→ difference between  $100\epsilon_f$  and  $100\epsilon_{pf}$  is small for ductile metals

→ see p.61 for other cases

%RA = percent reduction in area, another measure of ductility

- obtained by comparing the cross-sectional area after fracture,  $A_f$ , with the original area

$$\%RA = 100 \frac{A_i - A_f}{A_i}$$

$$\%RA = 100 \frac{d_i^2 - d_f^2}{d_i^2}$$

$U_f$  = tensile toughness, the area under the entire engineering stress-strain curve up to fracture

- measure of the ability of the material to absorb energy w/out fracture
- tensile toughness should not be confused w/ fracture toughness

$$U_f \approx \varepsilon_f \left( \frac{\sigma_0 + \sigma_u}{2} \right) \quad \begin{matrix} \text{if curve relatively flat} \\ \text{after yielding} \end{matrix}$$

Strain Hardening - rise in the stress-strain curve following yielding, as the material is increasing its resistance w/increasing strain

$$\text{strain hardening ratio} = \frac{\sigma_u}{\sigma_0} \quad \begin{matrix} > 1.4 & \text{= high} \\ < 1.2 & \text{= low} \end{matrix} \quad (\text{for metals})$$

$\tilde{\sigma}$  = true stress, the axial force divided by current x-sectional area

$$\tilde{\sigma} = \frac{P}{A} \quad \tilde{\sigma} = \sigma \frac{A_i}{A}$$

$$\tilde{\varepsilon} = \text{true strain} \quad \tilde{\varepsilon} = \ln(1 + \varepsilon)$$

THE CONSTANT VOLUME ASSUMPTION

$$\tilde{\sigma} = \sigma (1 + \varepsilon)$$

$$\tilde{\varepsilon} = \ln(A_i/A)$$

$$\tilde{\varepsilon} = 2 \ln\left(\frac{d_i}{d}\right)$$

Corrections for Triaxial Stress due to Necking - p. 81

- Bridgman
- Mirone

True Stress-Strain Curves

→ for metals in region beyond yielding, stress vs. plastic strain behavior fits relationship

$$\tilde{\sigma} = H \tilde{\varepsilon}_p^n$$

$$\uparrow \text{true plastic strain } \tilde{\varepsilon}_p = \tilde{\varepsilon} - \tilde{\sigma}/E$$

→ when plotted on log-log coordinates: get a straight line w/slope  $n$  (aka the strain hardening exponent)

→  $H$  = strength coefficient, intercept at  $\tilde{\varepsilon}_p = 1$

$\uparrow$  at large strains during advanced stages of necking, need to use true stresses that have been corrected w/Mirone (p. 81-83)  $\tilde{\sigma}_m = H \tilde{\varepsilon}_p^n$

$\tilde{\sigma}_f$  = true fracture strength

- obtained from force @ fracture + final area

$$\tilde{\sigma}_f = \frac{P_f}{A_f} = \sigma_f \left( \frac{A_i}{A_f} \right) \quad \dots \text{correction usually needed}$$

$\tilde{\varepsilon}_f$  = true fracture strain

- obtained from final area or percent reduction in area

$$\tilde{\varepsilon}_f = \ln\left(\frac{A_i}{A_f}\right) = \ln\left(\frac{100}{100 - \gamma_{RA}}\right)$$

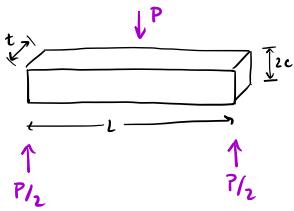
$\tilde{U}_f$  = true toughness

- area under true stress-strain curve up to fracture

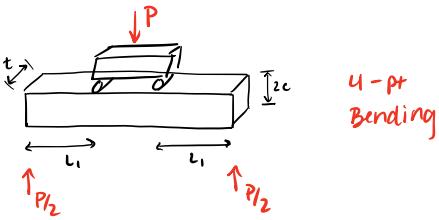
### Bending (Flexure) Tests - assume linear elastic behavior

$$\sigma = \frac{Mc}{I}$$

$M$  = bending moment



rectangular cross-section of depth  $2c$  and width  $t$ ,  
the area moment of inertia about neutral axis  
 $\therefore I = 2tc^3/3$



4-pt  
Bending

3-pt bending: highest bending moment occurs at midspan  $M = Pl/4$

$$\therefore \sigma_{fb} = \frac{3L}{8tc^2} P_f$$

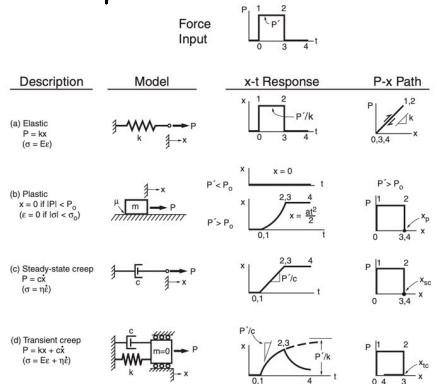
fracture strength

fracture force

$$\text{max deflection: } v = \frac{PL^3}{48EI}$$

$$\text{then } E = \frac{L^3}{48I} \left( \frac{dP}{dv} \right) = \frac{L^3}{32tc^3} \left( \frac{dP}{dv} \right)$$

### Models for Deformation Behavior



$$E = \frac{\sigma}{\epsilon} \quad \sigma = E\epsilon \quad \epsilon = \frac{\sigma}{E}$$

$$\text{let } P = Kx \rightarrow E = \frac{Kl}{A}$$

$$\text{for plastic deformation: } \sigma_0 = \frac{P_0}{A}$$

stress:  $\sigma = \eta_A$

strain:  $\epsilon = x/l$

$d\epsilon/dt = \dot{\epsilon} = \dot{x}/l$

Heady-state creep: proceeds at constant rate under constant force  $\Rightarrow \eta = \frac{\sigma}{\dot{\epsilon}}$  coefficient of tensile viscosity

transient creep: slows down as time passes

Plastic Deformation:  $\epsilon = \frac{\sigma}{E_1} + \frac{\sigma - \sigma_0}{E_2}$   $E_c = \frac{E_1 E_2}{E_1 + E_2}$

elastic, linear hardening

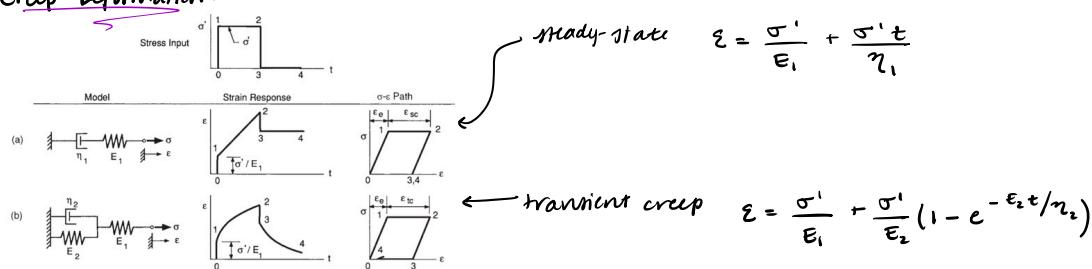


elastic, perfectly plastic



rigid, perfectly plastic

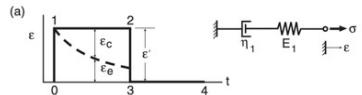
### Creep Deformation:



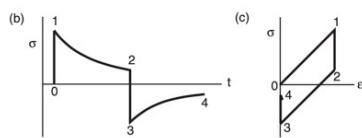
$$\text{steady-state } \epsilon = \frac{\sigma'}{E_1} + \frac{\sigma' t}{\eta_1}$$

$$\text{transient creep } \epsilon = \frac{\sigma'}{E_1} + \frac{\sigma' t}{E_2} (1 - e^{-\epsilon_2 t / \eta_2})$$

Relaxation - same phenomenon as creep, differing only in that it is observed under constant strain rather than constant stress



$$\sigma = E_1 \varepsilon' e^{-\eta_1 t / m_1}$$



↑ corresponds to a stress-time response that decays - that is, relaxes - with time (as illustrated by b.)



### Elastic Deformation

$$\nu = -\frac{\text{transverse strain}}{\text{longitudinal strain}} = -\frac{\varepsilon_y}{\varepsilon_x} \quad \dots \quad \varepsilon_y = -\frac{\nu}{E} \sigma_x$$

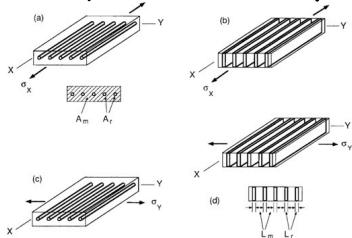
$$\begin{aligned} \varepsilon_x &= \frac{1}{E} [\sigma_x - \nu(\sigma_y + \sigma_z)] + \alpha(\Delta T) \\ \varepsilon_y &= \frac{1}{E} [\sigma_y - \nu(\sigma_x + \sigma_z)] + \alpha(\Delta T) \\ \varepsilon_z &= \frac{1}{E} [\sigma_z - \nu(\sigma_x + \sigma_y)] + \underbrace{\alpha(\Delta T)}_{\text{thermal term}} \end{aligned} \quad \left. \begin{array}{l} \text{Generalized Hooke's Law} \\ \left. \begin{array}{l} \gamma_{xy} = \frac{T_{xy}}{G} \\ \gamma_{yz} = \frac{T_{yz}}{G} \\ \gamma_{zx} = \frac{T_{zx}}{G} \end{array} \right. \end{array} \right\}$$

\*only 2 independent elastic constants are needed for an isotropic material, so that one of  $E$ ,  $G$ , and  $\nu$  can be considered redundant

$$G = \frac{E}{2(1+\nu)}$$

$$\text{Bulk Modulus } B = \frac{\sigma_n}{\varepsilon_n} = \frac{E}{3(1-2\nu)}$$

### Anisotropic - Fibrous Composites - $E$ parallel to Fibers



$$A = A_r + A_m$$

$$E_I = \frac{E_r A_r + E_m A_m}{A}$$

$$V_r = \frac{A_r}{A} \quad V_m = 1 - V_r = \frac{A_m}{A}$$

$$E_I = V_r E_r + V_m E_m$$

### Anisotropic - Fibrous Composites - $E$ transverse to Fibers p. 166

$$E_I = \frac{E_r E_m}{V_r E_m + V_m E_r}$$

### Other elastic constants

$$\text{Major Poisson's Ratio : } \nu_{xy} = V_r \nu_r + V_m \nu_m$$

$$\text{shear modulus : } G_{xy} = \frac{G_r G_m}{V_r G_m + V_m G_r}$$

$$\text{Plane Stress} \quad \sigma_z = \tau_{yz} = \tau_{zx} = 0$$

$$\sigma_1, \sigma_2 = \frac{\sigma_x + \sigma_y}{2} \pm \sqrt{\left(\frac{\sigma_x - \sigma_y}{2}\right)^2 + \tau_{xy}^2}$$

$$\tan 2\theta_n = \frac{2\tau_{xy}}{\sigma_x - \sigma_y} \quad \dots \text{coordinate axes rotations for maximum and minimum values of } \sigma$$

$$\tan 2\theta_s = -\frac{\sigma_x - \sigma_y}{2\tau_{xy}} \quad \dots \text{coordinate axes rotation for maximum shear stress}$$

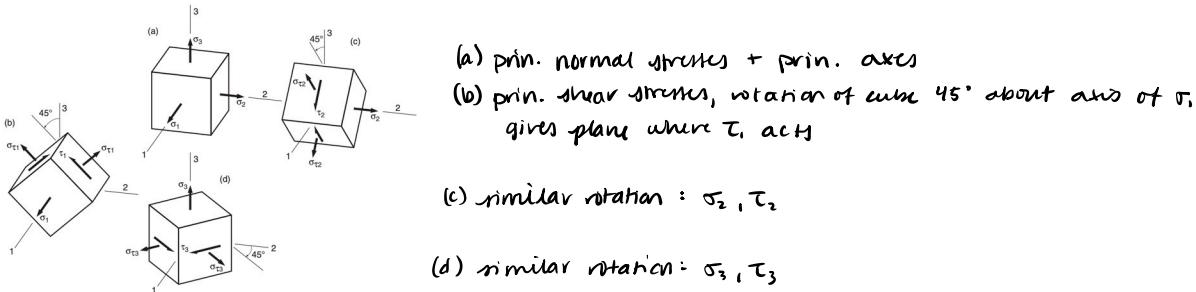
$$\tau_3 = \sqrt{\left(\frac{\sigma_x - \sigma_y}{2}\right)^2 + \tau_{xy}^2} \quad \dots \text{principal shear stress, aka maximum shear stress in } x, y\text{-plane}$$

↑ the 2 orthogonal planes where this shear stress occurs are found to have the same normal stress of

$$\sigma_{\tau_3} = \frac{\sigma_x + \sigma_y}{2}$$

$\tau_3$  and the accompanying normal stress can be expressed in terms of the principal normal stresses  $\sigma_1$  and  $\sigma_2$ :

$$\tau_3 = \frac{|\sigma_1 - \sigma_2|}{2} \quad \sigma_{\tau_3} = \frac{\sigma_1 + \sigma_2}{2}$$



The principal shear stresses are  $\tau_1, \tau_2, \tau_3$ , each accompanied by normal stresses that are the same on the two shear planes  $\sigma_{\tau_1}, \sigma_{\tau_2}, \sigma_{\tau_3}$

$$\tau_1 = \frac{|\sigma_2 - \sigma_3|}{2} = \frac{|\sigma_1|}{2} \quad \tau_2 = \frac{|\sigma_1 - \sigma_3|}{2} = \frac{|\sigma_1|}{2} \quad \tau_3 = \frac{|\sigma_1 - \sigma_2|}{2} = \frac{|\sigma_1|}{2}$$

$$\sigma_{\tau_1} = \frac{\sigma_1 + \sigma_2}{2}$$

$$\sigma_{\tau_2} = \frac{\sigma_1 + \sigma_3}{2}$$

$$\sigma_{\tau_3} = \frac{\sigma_1 + \sigma_2}{2}$$

when  $\sigma_2 = \sigma_{\tau_2}$   
 $= \tau_{zx} = 0$   
 then  $\sigma_3 = 0$

$$\tau_{\max} = \text{MAX}(\tau_1, \tau_2, \tau_3)$$

$\sigma_n$  = Octahedral normal stress aka hydrostatic stress

$$\sigma_n = \frac{\sigma_1 + \sigma_2 + \sigma_3}{3} \Rightarrow \sigma_n = \frac{\sigma_x + \sigma_y + \sigma_z}{3}$$

$\tau_n$  = octahedral shear stress

$$\tau_n = \frac{1}{3} \sqrt{(\sigma_1 - \sigma_2)^2 + (\sigma_2 - \sigma_3)^2 + (\sigma_3 - \sigma_1)^2} \Rightarrow \tau_n = \frac{1}{3} \sqrt{(\sigma_x - \sigma_y)^2 + (\sigma_y - \sigma_z)^2 + (\sigma_z - \sigma_x)^2 + 6(\tau_{xy}^2 + \tau_{yz}^2 + \tau_{zx}^2)}$$

principal normal and shear strains occur in similar manner as for stresses

→ plane strain:  $\varepsilon_z = \gamma_{yz} = \gamma_{zx} = 0$

$$\tan 2\theta_n = \frac{\gamma_{xy}}{\varepsilon_x - \varepsilon_y}$$

$$\varepsilon_1, \varepsilon_2 = \frac{\varepsilon_x + \varepsilon_y}{2} \pm \sqrt{\left(\frac{\varepsilon_x - \varepsilon_y}{2}\right)^2 + \left(\frac{\gamma_{xy}}{2}\right)^2}$$

$$\tan 2\theta_1 = - \frac{\varepsilon_x - \varepsilon_y}{\gamma_{xy}}$$

$$\gamma_3 = \sqrt{(\varepsilon_x - \varepsilon_y)^2 + (\gamma_{xy})^2}$$

$$\varepsilon_{\gamma_3} = \frac{\varepsilon_x + \varepsilon_y}{2}$$

3D states of Strain:

$$\begin{vmatrix} (\varepsilon_x - \varepsilon) & \frac{\gamma_{xy}}{2} & \frac{\gamma_{xz}}{2} \\ \frac{\gamma_{xy}}{2} & (\varepsilon_y - \varepsilon) & \frac{\gamma_{yz}}{2} \\ \frac{\gamma_{xz}}{2} & \frac{\gamma_{yz}}{2} & (\varepsilon_z - \varepsilon) \end{vmatrix} = 0$$

$$\gamma_1 = |\varepsilon_2 - \varepsilon_3|, \quad \gamma_2 = |\varepsilon_1 - \varepsilon_3|, \quad \gamma_3 = |\varepsilon_1 - \varepsilon_2|$$

For cases of plane stress,  $\sigma_z = \tau_{yz} = \tau_{zx} = 0$ , the Poisson effect results in normal strains  $\varepsilon_z$  occurring in the out-of-plane direction, so that the state of strain is 3D

→ if isotropic, no shear  $\gamma_{yz}$  or  $\gamma_{zx}$  occur then  $\varepsilon_z = \varepsilon$

$\varepsilon_z$  can be obtained from Hooke's Law applying  $\sigma_z = 0$ :

$$\sigma_x + \sigma_y = \frac{E}{1-\nu} (\varepsilon_x + \varepsilon_y)$$

$$\varepsilon_z = \frac{-\nu}{1-\nu} (\varepsilon_x + \varepsilon_y)$$

### Failure Criteria

general form:  $f(\sigma_1, \sigma_2, \sigma_3) = \sigma_c \text{ @ failure}$

↑ either yield strength  $\sigma_y$  or ultimate strength  $\sigma_u$  or  $\sigma_c$  depending on whether yield or fracture of interest

$$f(\sigma_1, \sigma_2, \sigma_3) = \bar{\sigma}$$

↑ effective stress

$$\begin{array}{ll} \bar{\sigma} = \sigma_c & \text{@ failure} \\ \bar{\sigma} < \sigma_c & \text{no failure} \end{array}$$

$$\downarrow \text{ safety factor} \\ X = \frac{\sigma_c}{\bar{\sigma}}$$

max normal stress fracture criterion → brittle materials (graphically rep. by square)

$$\bar{\sigma}_N = \max(|\sigma_1|, |\sigma_2|, |\sigma_3|)$$

fracture expected when  $\bar{\sigma}_N = \sigma_u$  (either tensile or compressive)

$$X = \frac{\sigma_u}{\bar{\sigma}_N}$$

max shear stress yield criterion → ductile materials

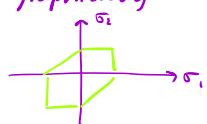
in uniaxial:  $\sigma_1 = \sigma_o, \sigma_2 = \sigma_3 = 0, \tau_o = \sigma_o/2$

$\therefore \sigma_o = \max(|\sigma_1 - \sigma_2|, |\sigma_2 - \sigma_3|, |\sigma_3 - \sigma_1|) \text{ @ yielding}$

$$\bar{\sigma}_s = \max(|\sigma_1 - \sigma_2|, |\sigma_2 - \sigma_3|, |\sigma_3 - \sigma_1|)$$

$$X = \frac{\sigma_o}{\bar{\sigma}_s}$$

graphically



octahedral shear stress yield criterion, von Mises, distortion energy criterion

\* predicts yielding occurs when shear stress on octahedral planes reaches crit. value

$$\bar{\sigma}_H = \frac{1}{\sqrt{2}} \sqrt{(\sigma_x - \sigma_y)^2 + (\sigma_y - \sigma_z)^2 + (\sigma_z - \sigma_x)^2 + 6(\tau_{xy}^2 + \tau_{yz}^2 + \tau_{zx}^2)}$$

graphically rep. by an ellipse