


RESEARCH ARTICLE | MAY 09 2016

Extended magnetic exchange interactions in the high-temperature ferromagnet MnBi **FREE**

T. J. Williams; A. E. Taylor ; A. D. Christianson; S. E. Hahn; R. S. Fishman; D. S. Parker; M. A. McGuire; B. C. Sales; M. D. Lumsden



Appl. Phys. Lett. 108, 192403 (2016)

<https://doi.org/10.1063/1.4948933>

 CHORUS



Journal of Applied Physics

Special Topic:

Disordered Materials at the Atomic Scale

Guest Editors: Jaeyun Moon, Matteo Baggioli

Submit Today!

Extended magnetic exchange interactions in the high-temperature ferromagnet MnBi

T. J. Williams,^{1,a)} A. E. Taylor,¹ A. D. Christianson,^{1,2} S. E. Hahn,³ R. S. Fishman,⁴ D. S. Parker,⁴ M. A. McGuire,⁴ B. C. Sales,⁴ and M. D. Lumsden¹

¹Quantum Condensed Matter Division, Neutron Sciences Directorate, Oak Ridge National Lab, Oak Ridge, Tennessee 37831, USA

²Department of Physics and Astronomy, University of Tennessee, Knoxville, Tennessee 37966, USA

³Neutron Data Analysis and Visualization Division, Neutron Sciences Directorate, Oak Ridge National Lab, Oak Ridge, Tennessee 37831, USA

⁴Materials Science and Technology Division, Physical Sciences Directorate, Oak Ridge National Lab, Oak Ridge, Tennessee 37831, USA

(Received 8 March 2016; accepted 27 April 2016; published online 9 May 2016)

The high-temperature ferromagnet MnBi continues to receive attention as a candidate to replace rare-earth-containing permanent magnets in applications above room temperature. This is due to a high Curie temperature, large magnetic moments, and a coercivity that increases with temperature. The synthesis of MnBi also allows for crystals that are free of interstitial Mn, enabling more direct access to the key interactions underlying the physical properties of binary Mn-based ferromagnets. In this work, we use inelastic neutron scattering to measure the spin waves of MnBi in order to characterize the magnetic exchange at low temperature. Consistent with the spin reorientation that occurs below 140 K, we do not observe a spin gap in this system above our experimental resolution. A Heisenberg model was fit to the spin wave data in order to characterize the long-range nature of the exchange. It was found that interactions up to sixth nearest neighbor are required to fully parametrize the spin waves. Surprisingly, the nearest-neighbor term is antiferromagnetic, and the realization of a ferromagnetic ground state relies on the more numerous ferromagnetic terms beyond nearest neighbor, suggesting that the ferromagnetic ground state arises as a consequence of the long-ranged interactions in the system. *Published by AIP Publishing.*
[\[http://dx.doi.org/10.1063/1.4948933\]](http://dx.doi.org/10.1063/1.4948933)

The unusual magnetic properties of MnBi produce a material that is an attractive candidate as an alternative to rare-earth-containing ferromagnets in applications at room temperature and above.¹ Its high transition temperature ($T_C = 630$ K) and strong magnetic anisotropy^{2–4} combined with large Mn moments ($3.50(2) \mu_B$ at 300 K and $3.90(2) \mu_B$ at 5 K (Ref. 5)) produce energy products up to 7.7 MGOe.⁴ Of particular interest for applications, MnBi has a magnetic coercivity that increases from ≈ 1 T at room temperature to ≈ 2.5 T at 550 K.^{6–8} It also displays favorable magneto-optical properties such as a large Kerr effect, which may be useful as a magnetic storage medium in thin films.^{9,10} For this reason, there is an ongoing, concerted effort to understand the growth conditions to optimize its properties for these applications.¹¹

Despite the recent interest in MnBi, the microscopic origin of its magnetic properties has not been completely determined. Above 140 K, the ordered magnetic moments lie along the c -axis, but between 140 K and 90 K the spins undergo a continuous rotation away from the c -axis so that below 90 K the spins lie in the ab -plane.⁴ This transition has been identified as a symmetry-lowering structural phase transition driven by magnetostriction.⁵ It has been suggested that the Bi atoms play an important role in driving this spin-reorientation, as the Bi-Bi exchange interactions have been proposed as an origin of the observed anisotropic thermal expansion.⁵ Local spin density approximation (LSDA) calculations suggest that the spin

reorientation occurs due to an inversion of the orbital momentum minimum on the Bi atom,¹² while a tight-binding atomic-sphere-approximation (ASA) model predicts a small, antiparallel moment on the Bi atom, which undergoes a simultaneous spin reorientation.⁴

The loss of magnetic order above 630 K is not a typical magnetic transition, but occurs instead due to a decomposition of MnBi into $Mn_{1.08}Bi$ and elemental Bi,¹³ both paramagnetic at this temperature. The product $Mn_{1.08}Bi$ has excess Mn, where the excess Mn atoms occupy 8% of the interstitial vacancies present in MnBi.¹³ The relative proximity to the eutectic point (535 K) makes the synthesis difficult, which is why most prior work has been performed on samples that are $Mn_{1.08}Bi$ and/or polycrystalline.^{3,4,8} Due to coupling of the magnetism with the structural decomposition, it has been found that application of a 10 T magnetic field can stabilize the MnBi phase up to 650 K.^{14–16} By cooling the melt in a magnetic field, MnBi crystallites can be obtained, as described elsewhere.^{5,15,17} Other isostructural Mn-based ferromagnets, such as MnSb,¹⁸ cannot be synthesized in this manner and so they contain some degree of interstitial Mn. Thus, studying the interstitial-free MnBi presents the opportunity to understand the magnetism and electronic correlations in the most fundamental form—coincidentally, the member of this family with the highest T_C and largest energy product.⁴

In this work, we use neutron scattering on a single crystal of MnBi that is free of interstitial Mn in order to measure

^{a)}williamstj@ornl.gov

the spin waves at low temperature. These measurements permit us to quantitatively determine the exchange interactions in this system, characterizing the degree of extended exchange that is present. Comparing this to other Mn-based binary magnets demonstrates the role that the long-ranged interactions play in the magnetic properties of this class of materials.

MnBi crystallizes in the hexagonal NiAs-type structure (space group $P6_3/mmc$), with the Bi atoms occupying half of the interstitial sites.^{3,4} For this study, single crystals of MnBi were grown in zero applied field from a flux utilizing excess Bi, as described elsewhere.⁵ The crystals have previously been characterized using powder and single-crystal x-ray diffraction measurements,⁵ whose refinements gave a nominal composition of $Mn_{1.01(1)}Bi$, where excess Mn atoms occupy interstitial sites. This can be seen in Fig. 1, where the open circles denote the interstitial positions that can be partially filled by the excess Mn atoms.

For the neutron scattering measurements, a single crystal of mass 1.989 g and a mosaic of 0.41° was rotated over a 90° range in 1° steps about a vertical axis, with the $[H\ 0\ L]$ plane horizontal. The measurements were performed on the Wide Angular-Range Chopper Spectrometer (ARCS) at the Spallation Neutron Source (SNS)¹⁹ using incident neutron energies of 80 meV and 150 meV, giving an energy resolution of 3.2 and 6 meV, respectively, at the elastic line. The sample was measured in a closed-cycle refrigerator at temperatures of 5 K and 200 K.

The measured magnetic Bragg peaks at 5 K and 200 K are consistent with moments directed perpendicular and parallel to the c -axis, respectively. The 5 K data, which have

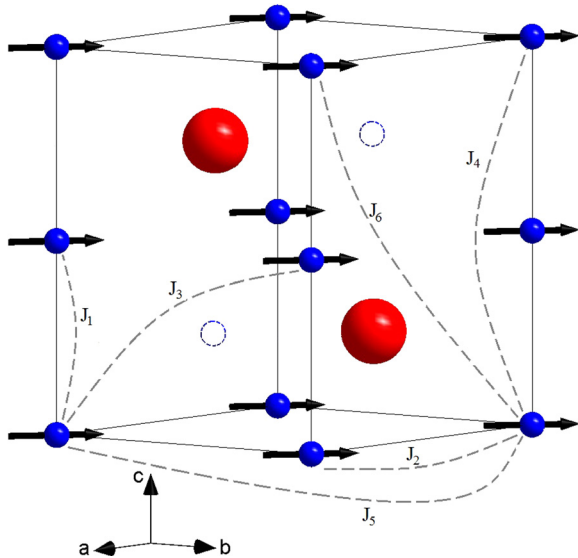


FIG. 1. The crystal structure of MnBi. The Mn atoms (small, blue) occupy the $2a$ position, while the Bi atoms (large, red) occupy the $2c$ position. The other half of the interstitial positions (Wyckoff symbol $2d$), shown as open circles, can be occupied by Mn impurities. Characterization measurements of the samples used suggested that 1(1)% of the interstitial sites were so occupied by Mn atoms.⁵ The spin arrangement shown is for the low-temperature range ($T < 90$ K). At all temperatures, MnBi is ferromagnetic; however above $T = 140$ K, the spins point along the c -axis, while from $T = 140$ K to 90 K, the spins rotate away from the c -axis such that below 90 K the spins lie entirely in the ab -plane. The exchange parameters (J_1 through J_6) determined in this work are shown between the relevant Mn ions.

reduced phonon scattering due to the Bose factor, were used to determine the magnetic exchanges in this system. The upper panels of Fig. 2 show representative slices through the spin wave spectrum measured on the ARCS spectrometer at $T = 5$ K, below the spin reorientation, so that the magnetic moments lie within the ab -plane.

The spin waves were measured across multiple directions in reciprocal space and are observed to be well-defined, allowing the exchange interactions to be quantified. In the case of $Mn_{1.13}Sb$ (shown in Fig. 2(e)), diffuse inelastic scattering is seen around the magnetic Bragg peaks that are attributed to the inclusion of interstitial Mn.¹⁸ The same measurement in MnBi (shown in Fig. 2(f)) does not show this diffuse component, consistent with the refinement of diffraction data that measured 1(1)% excess Mn in this sample. This allows for a simpler determination of the exchange interactions since we do not have to account for the effects of interstitial Mn on the magnetic exchange.

To parametrize the spin wave dispersion, the data were fit to a Heisenberg model. The moment was assumed to be $3.90 \mu_B$ on the Mn atoms,⁵ while no moment was placed on the Bi atoms. This is because the theoretical prediction for the size of the Bi moments is two orders of magnitude smaller than the Mn moments, and any moment on the Bi sites and accompanying Bi-Bi exchange interactions would be too weak to be observed in these measurements.^{4,12} In order to accurately reproduce the spin wave measurements, it was necessary to include interactions up to the 6th nearest neighbor (J_6), corresponding to a distance of 7.462 Å. These exchange interactions are shown schematically in Fig. 1. The need to include interactions up to $d = 7.5$ Å indicates the significance of relatively long-ranged interactions, likely due to an itinerant nature of the magnetic exchange. Similarly, other isostructural Mn binary systems MnP ,²⁰ $MnAs$,^{21,22} and $MnSb$ ^{18,23} all have large Mn moments and all order ferromagnetically at, or near, ambient pressure. Despite the symmetry-lowering transition at $T_{SR} = 90$ K, the high-temperature space group (NiAs-type) was used for the spin-wave calculations as previous measurements of the symmetry-lowering distortion indicated that the magnitude of the distortion is very small, and thus unlikely to significantly influence the magnetic excitation spectrum.⁵ For more details of the fitting procedure, see the supplementary material.²⁴

The magnetic anisotropy present in this system necessitates a non-zero spin gap. However, the possible origins of this anisotropy all result in a small gap,^{5,12} which agrees with the measurements of Fig. 2, where any spin gap is less than the resolution of the data, approximately 6 meV. Measurements on $MnSb$ at low temperatures likewise observed no gap, with a resolution of 0.5 meV.¹⁸ Moreover, the system undergoes a spin reorientation between 140 K and 90 K, suggesting that the magnetic anisotropy is small and strongly temperature-dependent.⁵ For these reasons, no spin gap term was included when fitting the data. The line through the data of Fig. 2 is the fit, showing that the dispersion was able to be accurately reproduced with the model that only included the Mn-Mn exchange terms given below, without any terms to account for Bi moments or a spin gap, though these may still be present. The lower panels of Fig. 2 show constant energy slices from the related material $Mn_{1.13}Sb$ at 20 meV,¹⁸ and from

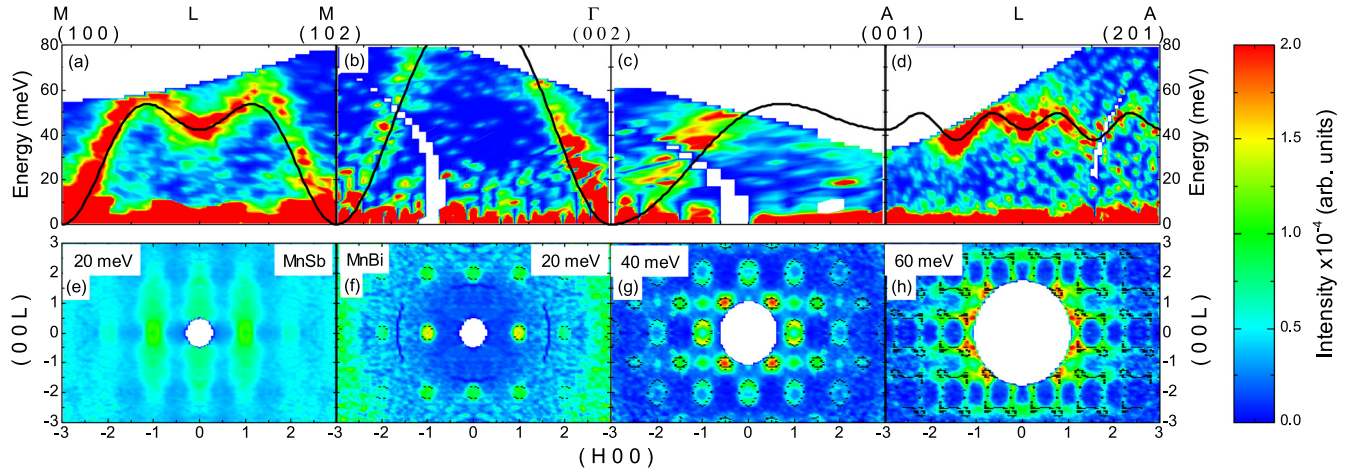


FIG. 2. Spin wave measurements and fits of MnBi at $T = 5$ K. (a)–(d) Inelastic neutron measurements using $E_i = 150$ meV along different reciprocal space directions in the $(H\ 0\ L)$ scattering plane. The line is a fit to a Heisenberg model, which required interactions up to J_6 . Bottom row: Slices of the MnBi data in the $(H\ 0\ L)$ scattering plane at 20 meV (f), 40 meV (g), and 60 meV (h), integrating the data ± 5 meV. The inelastic slices show the form factor dependence of the spin wave intensity, indicating their magnetic origin. (e) A slice of the related material $\text{Mn}_{1.13}\text{Sb}$ at 20 meV, multiplied by a factor of 5 to normalize it to the MnBi scale. (Data adapted with permission from Taylor *et al.*, Phys. Rev. B **91**, 224418 (2015). Copyright 2015 The American Physical Society.) Compared to panel (f), we see that there is additional diffuse scattering along L in $\text{Mn}_{1.13}\text{Sb}$, arising from the presence of interstitial Mn. The lack of similar elastic or inelastic signatures is consistent with the MnBi crystal studied being virtually free of interstitial Mn.

MnBi at 20 meV, 40 meV, and 60 meV. Comparing both 20 meV slices, we see that there is additional scattering in $\text{Mn}_{1.13}\text{Sb}$ along the L -direction, arising from the presence of interstitial Mn. The lack of that scattering in the MnBi data is consistent with the diffraction measurements that have only a small amount of interstitial Mn.⁵

The values of the exchange interactions are shown in Table I. The effective nearest-neighbor interaction is strongly antiferromagnetic, while the rest of the terms are smaller and ferromagnetic. However, due to the larger number of neighbors at greater distances, the ferromagnetic terms more than compensate for the J_1 term, leading to a ferromagnetic ground state. The antiferromagnetic J_1 term is somewhat unexpected, due to the large Mn-Mn distance,^{25,26} but a fit that forced J_1 to be ferromagnetic did not reproduce the data. As a further consistency check on the nature of the magnetic interactions, we have performed first principles calculations of the effective Heisenberg exchange parameters using the generalized gradient approximation (GGA)²⁷ in the planewave all-electron density functional theory code WIEN2K²⁸ (see the supplementary material²⁴). In agreement with the experiment, J_1 is found to be large and antiferromagnetic, but the ground state is ferromagnetic due to moderate ferromagnetic values for J_3 and J_5 . The remaining

exchange constants are all small, as was found in the fits to the experimental data. The calculations find that the antiferromagnetic state lies some 135 meV per Mn higher than the ferromagnetic ground state, leading to a mean-field estimated Curie temperature of 522 K, in reasonable agreement with experiment. While the nearest-neighbor exchange term being antiferromagnetic is unexpected, both the calculations and the experimental data conclude that the long-ranged nature of the Mn exchange gives rise to ferromagnetism. This competition between antiferromagnetic local exchange and the ferromagnetic itinerant exchange component may also be responsible for the reduction of the moment to $3.90\ \mu_B$ from the nominal moment of $gS = 4\ \mu_B$.⁵ However, the long-range character in MnBi clearly dominates, especially when compared to isostructural and isoelectronic Mn binary systems (see Table II).

Of these materials, MnBi has the highest Curie temperature and largest ordered moment. This is likely due to the larger inter-Mn distance, which reduces the nearest-neighbor exchange and places increased emphasis on the higher-order

TABLE I. The values for the exchange constants obtained from fitting the data in Fig. 2, where positive values denote antiferromagnetic exchange. In order to model the data, it was necessary to include up to 6th nearest-neighbor interactions due to the extended nature of the exchange.

Exchange	No. neighbors	Vector	Distance	Value (meV)
J_1	2	$c/2$	3.055 Å	4.70(17)
J_2	6	a	4.283 Å	-0.61(10)
J_3	12	$a + c/2$	5.261 Å	-1.73(3)
J_4	2	c	6.110 Å	-0.12(18)
J_5	6	$a + b$	7.418 Å	-1.29(8)
J_6	12	$a + c$	7.462 Å	-0.63(3)

TABLE II. Magnetic properties of various MnT ($T = \text{Bi, Sb, As, P}$) materials. These compounds are found to be ferromagnetic, where the distance between the Mn ions increases with increasing size of the T ion. This suggests that there is a systematic increase in the role of the extended exchange, which results in higher moments and transition temperatures for larger T atoms in the MnT series. This agrees well with the results of the inelastic neutron scattering on MnBi.

Compound	T_{Curie} (K)	Moment (μ_B)	NN Mn Dist. (Å)
MnBi ⁵	630	3.90	3.055
MnSb ^{29,30}	587	3.55	2.895
MnAs ^{22,31}	318	3.20	2.852
MnP ^{20,32,a}	292	1.33	2.743

^aThe compound MnP is orthorhombic, a slight distortion of the NiAs-type structure of the other compounds listed here. This leads to a helical magnetic state below 50 K; however, the magnetic structure between 50 K and $T_C = 292$ K is the same as for the other MnT compounds.

exchange terms. Therefore, we expect similar spin wave measurements of these compounds to demonstrate a systematic shift from long-ranged to more local interactions. These measurements are greatly complicated by the inclusion of interstitial Mn,¹⁸ and it also remains an open question to what degree the interstitial ions influence the moment and exchange pathways in these various systems.³⁰ However, the systematic changes through the Mn binary ferromagnets are likely controlled by the varying importance of the long-range interactions.

In conclusion, we have observed well-defined spin waves in MnBi. In order to characterize the spin waves, a Heisenberg model with six exchange constants has been fit to the data. The need to include terms out to $d = 7.5$ Å suggests that the Mn-Mn exchange is itinerant, and the data taken here show that any gap in this material is less than 6 meV. The resulting fits to the data have shown that the nearest-neighbor term is antiferromagnetic, while all of the higher-order terms are ferromagnetic. This quantitative determination of the exchange parameters shows that the long-range interactions are the determining factor in the ferromagnetic ordering, a trend that can explain the reduction in ordered moment and Curie temperature in isostructural and isoelectronic Mn binary systems.

The strength and sign of the exchange constants in MnBi should form a basis for comparing to non-stoichiometric Mn_{1.08}Bi, as well as other Mn binary ferromagnets: MnAs, MnP, and MnSb, particularly in understanding the role of interstitial Mn on the magnetic properties. Finally, the results obtained here shed light on the role of extended exchange in determining the essential characteristics of MnBi for use in permanent magnet applications.

We acknowledge instrument support from D. A. Abernathy and J. Niedziela. This research at ORNL's Spallation Neutron Source was sponsored by the Scientific User Facilities Division, Office of Basic Energy Sciences, U.S. Department of Energy. T.J.W. acknowledges the support from the Wigner Fellowship program at Oak Ridge National Laboratory. M.A.M. acknowledges the support from U.S. Department of Energy, Office of Energy Efficiency and Renewable Energy, Vehicle Technologies Office, Propulsion Materials Program. B.C.S. and D.S.P. were supported by the Critical Materials Institute, an Energy Innovation Hub, funded by the U.S. Department of Energy, Office of Energy Efficiency and Renewable Energy, Advanced Manufacturing Office. R.S.F. was supported by the Department of Energy, Office of Science, Basic Energy Sciences, Materials Sciences and Engineering Division. This manuscript has been authored by UT-Battelle, LLC under Contract No. DE-AC05-00OR22725 with the U.S. Department of Energy. The United States Government retains and the publisher, by accepting the article for publication, acknowledges that the United States Government retains a non-exclusive, paid-up, irrevocable,

world-wide license to publish or reproduce the published form of this manuscript, or allow others to do so, for United States Government purposes. The Department of Energy will provide public access to these results of federally sponsored research in accordance with the DOE Public Access Plan (<http://energy.gov/downloads/doe-public-access-plan>).

- ¹N. Poudyal and J. P. Liu, *J. Phys. D: Appl. Phys.* **46**, 043001 (2013).
- ²F. Heusler, *Z. Angew. Chem.* **17**, 260 (1904).
- ³B. W. Roberts, *Phys. Rev.* **104**, 607 (1956).
- ⁴J. B. Yang, W. B. Yelon, W. J. James, Q. Cai, S. Roy, and N. Ali, *J. Appl. Phys.* **2002**, 7866 (2002).
- ⁵M. A. McGuire, H. Cao, B. C. Chakoumakos, and B. C. Sales, *Phys. Rev. B* **90**, 174425 (2014).
- ⁶T. Chen and W. E. Stutius, *IEEE Trans. Mag.* **10**, 581 (1974).
- ⁷X. Guo, X. Chen, Z. Altounian, and J. O. Ström-Olsen, *Phys. Rev. B* **46**, 14578 (1992).
- ⁸J. B. Yang, Y. B. Yang, X. G. Chen, X. B. Ma, J. Z. Han, Y. C. Yang, S. Guo, A. R. Yan, Q. Z. Huang, M. M. Wu, and D. F. Chen, *Appl. Phys. Lett.* **99**, 082505 (2011).
- ⁹H. J. Williams, R. C. Sherwood, F. G. Foster, and E. M. Kelley, *J. Appl. Phys.* **28**, 1181 (1957).
- ¹⁰D. J. Sellmyer, R. D. Kirby, J. Chen, K. W. Wierman, J. X. Shen, Y. Liu, B. W. Robertson, and S. S. Jaswal, *J. Phys. Chem. Solids* **56**, 1549 (1995).
- ¹¹J. Cui, J.-P. Choi, E. Polikarpov, M. E. Bowden, W. Xie, G. Li, Z. Nie, N. Zarkevich, M. J. Kramer, and D. Johnson, *Acta Mater.* **79**, 374 (2014).
- ¹²V. P. Antropov, V. N. Antonov, L. V. Bekenov, A. Kutepov, and G. Kotliar, *Phys. Rev. B* **90**, 054404 (2014).
- ¹³T. Chen, *J. Appl. Phys.* **45**, 2358 (1974).
- ¹⁴E. Adams, *Rev. Mod. Phys.* **25**, 306 (1953).
- ¹⁵Y. Liu, J. Zhang, S. Cao, X. Zhang, G. Jia, Z. Ren, X. Li, C. Ling, and K. Deng, *Phys. Rev. B* **72**, 214410 (2005).
- ¹⁶T. Onogi, K. Koyama, and K. Watanabe, *J. Jpn. Inst. Met.* **71**, 489 (2007).
- ¹⁷H. Morikawa, K. Sassa, and S. Asai, *Mater. Transitions JIM* **39**, 814 (1998).
- ¹⁸A. E. Taylor, T. Berlijn, S. E. Hahn, A. F. May, T. J. Williams, L. Poudel, S. Calder, R. S. Fishman, M. B. Stone, A. A. Aczel, H. B. Cao, M. D. Lumsden, and A. D. Christianson, *Phys. Rev. B* **91**, 224418 (2015).
- ¹⁹D. L. Abernathy, M. B. Stone, M. J. Loguillo, M. S. Lucas, O. Delaire, X. Tang, J. Y. Y. Lin, and B. Fultz, *Rev. Sci. Instrum.* **83**, 015114 (2012).
- ²⁰H. Obara, Y. Endoh, Y. Ishikawa, and T. Komatsubara, *J. Phys. Soc. Jpn.* **49**, 928 (1980).
- ²¹N. Menyuk, J. A. Kafalas, K. Dwight, and J. B. Goodenough, *Phys. Rev.* **177**, 942 (1969).
- ²²S. Gama, A. A. Coelho, A. de Campos, A. M. G. Carvalho, F. C. G. Gandra, P. J. von Ranke, and N. A. de Oliveira, *Phys. Rev. Lett.* **93**, 237202 (2004).
- ²³C. Guillard, *Ann. Phys. (Paris, Fr.)* **4**, 671 (1949).
- ²⁴See supplementary material at <http://dx.doi.org/10.1063/1.4948933> for a detailed description of the fitting procedure and first principles calculations used in this work.
- ²⁵R. Forrer, *Ann. Phys.* **7**, 605 (1952).
- ²⁶V. Seshu Bai and T. Bajasekharan, *J. Magn. Magn. Mater.* **42**, 198 (1984).
- ²⁷J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996).
- ²⁸P. Blaha, K. Schwarz, G. Madsen, D. Kvasnicka, and J. Luitz, *WIEN2k, An Augmented Plane Wave + Local Orbitals Program for Calculating Crystal Properties* (Techn. Universität Wien, Austria, 2001).
- ²⁹K. Elankumaran, G. Markandeyulu, and K. V. S. Rama Rao, *J. Phys. Soc. Jpn.* **61**, 1979 (1992).
- ³⁰P. Radhakrishna and J. W. Cable, *Phys. Rev. B* **54**, 11940 (1996).
- ³¹M. Yuzuri and M. Yamada, *J. Phys. Soc. Jpn.* **15**, 1845 (1960).
- ³²A. Continenza, S. Picozzi, W. T. Geng, and A. J. Freeman, *Phys. Rev. B* **64**, 085204 (2001).